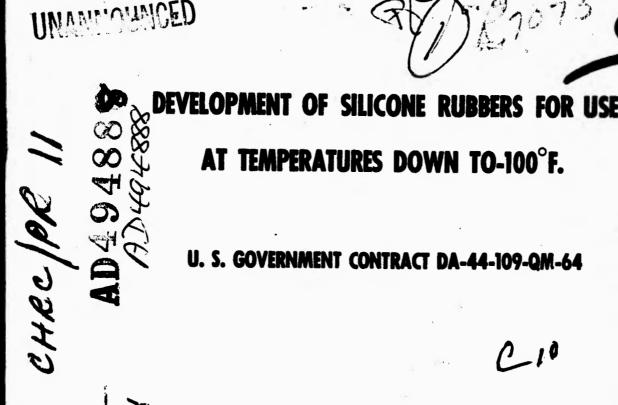
UNCLASSIFIED

AD NUMBER: ADO	0494888						
LIMITATION CHANGES							
TO:							
Approved for public release; distribution is unlimited.							
FROM:							
Distribution authorized to US Government A	gencies and their Contractors;						
Administrative/Operational Use; 5 Jun 1952.	Other requests shall be						
referred to US Army Materiel Command, Red	dstone Arsenal, AL 35898						
Aberdeen Proving Ground, MD 21010-5423.							

AUTHORITY

USAMC ltr dtd 19 Jun 1969



PROGRESS REPORT

FOR THE PERIOD DECEMBER 5, 1951 to JUNE 4, 1952

REPORT No. 11

PBD

1.5.

LIBRARY OF CONGRESS REFERENCE DEPARTMENT TECHNICAL IN UNMATION DIVISION FURMERLY (NAVY RESEARCH SECTION)

OCT 8.

0_10

CONNECTICUT HARD RUBBER COMPANY

WINNIL LIB

NEW HAVEN, CONNECTICUT

FEB 28 1969



THE ARMED SERVICES TECHNICAL INFORMATION AGENCY 1400 PENNSYLVANIA AVE., N.W., WASHINGTON 25, D.C.

> Tolophone: Liberty 5-6700 Ext: 6-6281

JUN 1 1 1954

MEMORANDUM FOR CHIEF, ASTIA REFERENCE CENTER

SUBJECT: Declassification of Documents

Reference (a) AFR 205-1, 15 Dec 53 (Executive Order 10501)

1. Beginning 15 June 1954, all items in ASTIA custody bearing obsolete classification "RESTRICTED" and not specifically upgraded in accordance with reference (a), will be regarded as unclassified in accordance with reference (a).

2. Appropriate markings will be made to effect changes in classification.

my R. Barner

ALBERT E. HINDMARSH Captain, USN Deputy Director, Security

THE CONNECTICUT HARD RUBBER COPPANY NEW HAVEN, CONNECTICUT 8887677Ú 16 DEVELOPMENT OF SILICONE PUBBERS FOR USE AT TEMPERATURES DOWN TO -100 DA-44-109-01-64 U. S. GOVERNMENT CONTRACT Progress Repet. E 5 Decommente 151-4 Jun 152 GF. L. /Kilbourne, Jr., W. / Stubble bine C. A. / No lken Juni 🏨 🗰 52

044 800

TABLE OF CONTENTS

5

															•	Page No.
List o	of Tables	••	••	•	• •	•	•	•	•	••	•	•	•	٠	•	111
List o	of Figure	S	• •	•	• •	٠	•	•	•		٠	•	•	٠	•	iv
Summar	ry of Res	ults	• •	٠	••	٠	•	•	•	• •	•	•	•	٠	•	vi
I. Mol	ecular W	eight	and	1 5	wel	lin	œ	st	nga	69						
on	Silicone	Rubb	er.	•	• •	•	•	•	• •		٠	•	•	•	•	1
Α.	Introdu	ction	• •	•	• •	٠	•	•	• •	•	•	٠	•	•	٠	1
В.	Experim	ental	••	•	• •	•	٠	•	• •	•	•	•	•	•	٠	5
	Test 3,	Effe on S	ct o well	of Lin	Ben g Ci	zoy ha r	l ac	Pe: te:	roy ris	id tid	e] Cs	Loa	ldi	ing	3	5
	Test 4,													•	•	-
	1830 4,	Swell	ling	ς C	har	act	er	is	tic	5101 S	19 •	or •	•	٠	٠	12
	Test 6,	Swell Vario	l of ous	So So	ili lve	con nts	e	E14 •	est	ome •	er •	ir •	•	•	•	25
C.	Discuss	ion .	•	•	• •	•	•	•	• •	•	•	•	•	•	•	45
Person	nel List		•	•	• •	•		•		•	•	•	•	•		47

LIST OF TABLES

Table	Pr	age	No.
A-1	Program of Swelling Study	3	
A-2	Compendium of Test Conditions and Results .	4	
B-1	Sample Preparation for Test 4	18	
B-2	Rate of Swell Equations for Silicone Rubber in Various Solvents	35	
B-3	Calculations of the Molecular Weight Between Cross-Links and the Polymer- Solvent Interaction	44	

111

LIST OF FIGURES

F

Figure		Pe	age No.
B-1	Swell of Silicone Rubber in Benzene at 25°C. (Varied Benzoyl Peroxide Content)		-
B-2	Press Cure Conditions for Test 4		
B-3	Oven Cure Cycle for Test 4f	• •	20
в-4	Swell of Silicone Rubber in Benzene (Varied Press Cure Conditions)		21
B-5	Swell of Silicone Rubber in Benzene (Varied Oven Cure Conditions)	•	22
в-6	Determination of "Equilibrium Swell , (Veried Press Cure Conditions)	•	23
B-7	Determination of "Equilibrium" Swell (Varied Oven Cure Conditions)		
B-8	Swell of Silicone Rubber in Various Solvents (Toluene, Cyclohexane, n-Octane, Methylcyclo- hexane, and d-Limonene)	•	33
B-9	Swell of Silicone Rubber in Various Solvents (Benzene, Xylene, n-Hexarie and Methyl Ethyl Ketone)	•	34
B-10	Determination of Equilibrium Swell of Silicone Rubber in Various Solvents (Toluene, Cyclohexan n-Octane, Methycyclohexane, and d-Limonene)	•	
	Determination of Equilibrium Swell of Silicone Rubber in Various Solvents (Benzene, Xylene, n-Hexane and Methyl Ethyl Ketone)		-
B-12	Relation of Swell to the Solubility Parameter of the Solvent		
B-13	Determination of the Solubility Parameter of the Polymer, Method of G. Geg		-

iv

LIST OF FIGURES (Cont.)

Figure

F

1.2

Page No.

Y

в-14	Determination of the Solubility Parameter of the Polymer by Approximation of the Flory Equation
B-15	Relation of the Polymer-Solvent Interaction Coefficients to Thermodynamic Functions of the Polymer and Solvent
B-16	Correlation of the Rate of Swell of Silicone Rubber with the Liquid Diffusivity of the Solvent
B-17	Correlation of the Rate of Swell of Silicone Rubber with the Degree of Equilibrium Swell . 43

SUMMARY OF RESULTS

Note: This reports covers the contract period from December 5, 1957, to June 4, 1952, but) refers exclusively to that portion of the research which is concerned with a study of the mechanism of swelling of silicone rubber. Report No, 12 will cover the same period of time and all other work done under this contract during this period. Results previously reported in Report No. 10 which had to do with the early swelling experiments are repeated below for the convenience of the reader.

From Report No. 10:

1. Silicone rubber swells rapidly in benzene and other good solvents during the first 10 hours at 25°C. After this time, swelling continues at a gradual rate up to at least 1000 hours and shows no indications of reaching a stationary state. This continued incremental swelling is attributed to the breaking of the molecular structure by thermal agitation of the swollen molecular structure.

2. Extrapolation of this portion of the swelling curve to zero time gives values of v_2 , the volume fraction of polymer in the swollen network. This thermodynamic quantity can be used to calculate M_c , the molecular weight between cross-links of the vulcanized polymer.

3. The swelling measurements were found to be affected by the simultaneous extraction of polymer by the swelling liquid. The swelling measurements were corrected for this extraction, this being the first time as far as we can learn from the literature, that swelling measurements have been thus corrected. In the case of a very lightly vulcanized sample bits correction is absolutely essential since the sample bits to de-swell after about 30 hours in the solvent. The percentage of polymer that is extractable decreases with increasing concentration of curing agent as does also the degree of swelling. These changes are proportional to the concentration of benzoyl peroxide below 2 per cent but there is very little difference in the polymers which are cured with 2 and 8 per cent benzoyl peroxide.

4. A completely novel method of measuring swelling continuously was devised and the equipment was designed, constructed and used in this work.

5. Data have been accumulated for calculation of M_c after the determination of μ_i , the interaction coefficient between solvent and polymer.

6. Experimental work on swelling has been correlated with the theoretical equations developed by Flory, Huggins, Gee and others who have studied swelling by following thermodynamic principles.

From the present report:

7. Incremental swelling was observed with all of the solvents used except with perfluoro(methylcyclohexane) which did not swell the silicone polymer at all and

vii

methyl ethyl ketone which was the poorest swelling agent of the nine remaining solvents. The statement in result No. 1 above, that swelling "shows no sign of reaching the stationary share," must be qualified by saying that it probably down not supply to poor swelling agents.

8. In Test 3, the effect of varying the benzoyl peroxide curative from 0.5 to 2.0 and 8.0, was explored by means of the swelling measurements. Exact values for M_c , the molecular weight between cross-links, cannot be calculated from the data since the value of \mathcal{M} , the polymer-solvent interaction coefficient, for each pair of rubber and solvent, has never been determined. It was possible, however, to place maximum and minimum limits on M_c and then to average these values arriving at reasonable average values of M_c . These are given in the table on the following page.

The values of M_C show clearly that the first 2 per cent of benzoyl peroxide added to a silicone recipe is far more efficient in its cross-linking action then the next 6 per cent, The average M_C with 2 per cent of benzoyl peroxide was 11,000 and with 8 per cent was 9,000. 9. In Test 4 wide variations in the degree of vulcanization of silicone rubber were made and swelling tests again carried out. The resulting maximum and minimum M_C values were again averaged and the results are given in the table.

viii

Variation of M_c with the Method of Preparation of Silicone Rubber#

	Per cont	<u>1</u> rea	e Cure	Oven			
Test	Benzoy Peroxide	Time Min	^m € nip , ^o F .	Time Hr.	Temp. oF.	M _c	
3a 3c 3i	0.5 20 8.0	15 15 15	250 250 250	24 24 24	300 300 300	>24,000 11,000 9,000	
4a 46 46 40 46 4f	2.0 2.0 2.0 2.0 2.0 2.0	<u>000000000000000000000000000000000000</u>	250 200 200 200 250	24 24 24 <u>24</u> <u>1</u> 48-hr to h	300 300 <u>300</u> <u>300</u> . cycle 80°F.	17,000 14,000 18.000 11,000 24,000 13,000	

*Important variables underlined.

10. The average values of M_c indicate that a 1-hour oven cure at 300°F. produces enough cross-links to bring the molocular weight down to 24,000, while a 24-hour cure at 300°F. reduces it to from 11,000 to 18,000. A gradually rising temperature cycle extending over 48 hours at a maximum temperature of 480°F. had the effect of reducing M_c to 13,000 in a single test.

11. It is known that the primary vulcanization reaction with benzoyl peroxide is completed in the initial press cure or in the very early stages of the oven cure. Since further cross-linking is evident during prolonged oven curing, it is suggested that loss of chain ends during the oven cure results in further cross-linking.

12. Perfluoro (mothylcyclohexane) does not swell silicone rubber within experimental error.

13. In increasing order of degree of swelling at

equilibrium, the solvents used are listed below, together with the equilibrium v_2 values:

Test	Solvent	v ₂ equil.
6j	Perfluoro(methylcyclo- hexane)	1.0
6h	Methyl ethyl ketone	. 285
6 i	Benzene	. 206
6g	d-Limonene	.183
6c	Xylene	.182
6a	Toluene	.170
6a	Cyclohexane	.148
60	n-Octane	•141
6b	n-Hexane	.138
6 f	Methyl cycl ohexane	.137

14. Since the solvents in the above list were compared for swelling activity with a single sample of cured rubber in which M_c was, therefore, constant, it was possible to arrive at relative values of μ , the solvent-polymer interaction coefficient, in each case. By assuming certain maximum and minimum values for M_c , corresponding maximum and minimum values for μ were calculated. From these values a relationship between μ and $V_1 \& 1$ was plotted. Thermodynamic considerations by Gee and Flory and others predict the existence of this relationship between μ , which is related to the heat of mixing, and $\$_1$, which is the cohesive energy density. This relationship is plotted on page μ .

X

15. An improvement in Gee's method of calculating the value of δ_2 , coherive energy density of the rubber, has been worked out. By this method, the cohesive energy density of vulcanized silicone rubber (General Electric SE-76) has been found to be 7.50 $\frac{1}{2}$.05.

16. No correlation between the rate of swelling of silicone rubber and the liquid diffusivity, or between the rate of swelling and the ultimate value of v_2 was found. 17. The polymer-solvent interaction coefficient \mathcal{H} (Flory) does not correspond to the similar function found in osmotic pressure equations (Huggins), at least for polydimethylsiloxane and methyl ethyl ketone.

18. It is suggested in the Discussion how it will be possible to arrive exactly at values of M_c and μ . It is also suggested that swelling measurements will serve as a tool to investigate the mechanism of reinforcement. It is obvious, also, that this tool will be useful in investigating the curing action of certain surface-active pigments. 19. It is felt that this work is an important contribution to the theory of structure of vulcanized elastomers.

xi

I. MOLECULAR WEIGHT AND SWELLING STUDIES ON SILICONE RUBBER

A. Introduction

The program of this swelling study was outlined in a series of eight experiments which was presented in Table IV-A-1, page 75 of Progress Report 10. This same outline is presented in Table A-1, page 3 of this report.

Tests 1, 2, 3 and 8 were reported in the previous Progress Report; and a review of Test 3 along with Tests 4 and 6 is reported in the following sections. These tests bear considerable practical as well as theoretical importance and will provide a sound basis for further investigations on the nature of pigment reinforcement.

Future work will be directed toward Test 7, Determination of the Effect of Pigment Reinforcement on the Rate and Magnitude of Swell of Silicone Rubber, in an effort to learn more about the practical as well as the theoretical mechanism of pigment reinforcement.

The apparatus for volume measurement was described in Section IV-B, pp. 76-82, of Progress Report 10, and the procedure was varied only slightly for the subsequent tests during the period this report covers. This variation involved the weighing of swollen rubber samples within a tare bottle containing the proper solvent; frequent gravimetric measurements and volume calculations therefrom thus served as a check on the accuracy of the

volumetric measurements and provided accuracy to 1 per cent.

4 - ¹.

1

A compendium of all test conditions and results is presented in Table A-2, page 4.

TABLE -A-1

Program of Swelling Study

Test		Object	Conditions
1		swelling measurement.	Silicone polymer. No pigment. 2% benzoyl peroxide.
	b.	Investigation of size of sample in relation to the rate of swell and the precision of measurement.	Benzene solvent. Varied sample size.
2	a. b.	Investigation of a secondary swell. Determination of the existence of an equilibrium swell.	Repeat swell of samples of Test l in benzene.
3		Determination of the effect of benzoyl peroxide concentration on the swell of silicone rubber.	Silicone polymer. No pigment. Varied benzoyl peroxide. Constant cure. Benzene solvent.
4.		Determination of the effect of curing conditions on the swell of silicone rubber.	Silicone polymer. No pigment. 2% benzoyl peroxide. Varied cure. Benzene solvent.
5.		Determination of the effect of heat treatment of silicone polymer prior to compounding.	Silicone polymer heat treated. No pigment. 2% ben- zoyl peroxide. Benzene solvent.
6.		Determination of the effect of various solvents and calculation of heat and entropy of mixing, Attempt to correlate thermodynamic properties with swelling results.	Silicone polymer. No pigment. 2% benzoyl peroxide. Constant cure. Varied solvents.
7.		Determination of the effect of pigment reinforcement on the rate and magnitude of swell of silicone rubber.	Silicone polymer. Varied pigments, pigment loadings. 2% benzoyl peroxide. Constant cure. Benzene solvent.
8		Determination of the rate of ex- traction of silicone polymer from vulcanized network by solvent action.	Silicone polymer. No pigment. Varied benzoyl peroxide. Varied cure. Benzene solvent.

COM

Test	Variable Studied	S a m p Formula parts b.perox.	Cu Time	ring C	onditic Time	ons	l o n Sise in. x.070"	Solv
l b c d f	Sample Size	2.0	15	250	24	300	$\frac{3}{8} \times \frac{1}{4}$ 9/16 × 1/4 3/4 × 1/4	Bense 4
	Secondary St							Benze
a b c d			lc aft ld le lf	er 238	-hour s	well ar	nd de-swell	
3	Compounding	formula	15	250	24	300		Bens
* 1 * 2		0.5					1/2 x 3/10	5
°1°2		2.0					3/4 x 1/4	
i 1 i 2		8.0					3/4 x 1/4	
4	Curing Condi	ltions					3/4 x 1/4	Benz
a1#2		2.0	10	250	24	300	.,, -	
b1b2	•		10	230	24	300		
°1°2			15	230	24	300		
d1d2			15	250	24	300		
•1•2			15	250	1	300		
f ₁ f ₂			15	250	48 t	o 480°F	. gradual 1	ise
6 •1•2		2.0	15	250	24	30 0	3/4 x 1/4	Tolu
b 1 b 2								n-Hei
°1°2							-	Iyle
d1d2							C	yoloh
•1•2								n-001
$f_1 f_2$							Methylo	-
8182								d-Lim
h ₁ h ₂							Methyl eth	-
								-
1112 1112 1112							ro(methylcy	Benz

1

/

COMPENDIUM OF TEST CONDITIONS AND RESULTS

-

olvent	Molal Vol. C.C.	Solubility Parameter	Duratic of Swell hr.	on v ₂ equil	M _C L. max.	M c min.	M NAX.	H min.
enzene	89.40	9.16	238					
ensene	89.40	9.16	1012					
nsene	89.40	9.16						
		•	174	.084	1,000,000	24,200	.521	.436
		•	144	. 206	14,900	6560		
		-	218	.229	12,000	5400		
nzene	89.40	9.16					.521	
			J 146	.187	24,200	8970	•741	.436
			145	.205	20,500	7600		
			146	.193	25,900	8840		
			بلبلا	.206	14,900	6560		
			J142	.180	38,000	10,900		
			145	.195	19,300	7830		
luene	106.85	8 03	1		14,900	6560		
Hexane		8.91	174	.170			•439	.295
lene	131.60	7.27	<u>144</u>	.138			• 323	.068
bexane	123.46	8.82	որի	.182			.458	.311
Octane	108.74	8.20	זויור	8بلا .			.400	.211
	163.52	7.55	175	.141			. 308	.000
hexane Monene	128.33	7.83	144	.137			•333	.071
laonene letone	162.2	8.5	144	.183			.416	.224
isene	90.16	9.22	յին	.285			•577	.531
exane)	89.40	9.16 6 r	144	.206			.521	.436
MADIE)	173.3	6.5	145	1.00				

2

B. Experimental

Test 3 - Effect of Benzoyl Peroxide Loading on Swelling Characteristics

<u>Object</u>: This test was for the purpose of determining the effect of vulcanization with varied benzoyl peroxide concentration on the swelling characteristics of silicone polymer, and thus by calculation of M_c to substantiate authentically the decreasing effectiveness of benzoyl peroxide as a cross-linking agent when used in concentrations above the optimum of about 2.0 parts per 100 parts of polymer.

5.

<u>Procedure</u>: Test samples 3a, 3c and 3i were prepared from slabs compounded with 0.5, 2.0 and 8.0 parts of benzoyl peroxide per 100 parts of silicone polymer. Milling conditions were similar, and slabs were press cured 15 minutes at $250^{\circ}F$. and oven cured 24 hours at $300^{\circ}F$.

Samples were immersed in pure benzene at 25.0°C. and volumes were measured at frequent intervals. After completion of the test, samples were de-swelled and weighed.

Calculations of v₂ were made by the same procedure as discussed on page 85 of Progress Report 10. <u>Results</u>: The results of Tests 3a and 3i were presented in Figures IV-C-4 and IV-C-5 following page 92 of Progress Report 10. The swelling of sample 3c was similar to that of Test 1 (samples were prepared similarly) but the present results are more accurate. Distinctly linear log-log relationships of v₂ (the volume fraction of polymer in the swollen network) with time were obtained as in previous tests with a gradual transition between the initial and incremental periods. The rate as well as the magnitude of swell varied considerably with the benzoyl peroxide content, as is shown by the following equations of the linear curves:

	Initial Period	Incremental Period
Test 3a	.083 to 4.0 hours	25 to 175 hours
	$\ln v_2 = -,496 \ln(t) -1.368$	$\ln v_2 =0254 \ln(t) -2.407$
Test 3c	.083 to 1,5 hours	6 to 145 hours
	$\ln v_2 =310 \ln(t) -1.194$	$\ln v_2 =0186 \ln(t) -1.537$
Test 3i	.083 to 2.0 hours	10 to 230 hours
	$\ln v_2 =257 \ln(t) -1.036$	$\ln v_2 =0521 \ln(t) -1.236$

Equilibrium swell as defined by J. R. Scott was determined in Figure B-1, page 11, (replacing Figure IV-C-6 of Report 10); from these values and from the molal volume of benzene and the density of polymer, equations for M_c in terms of μ were calculated by means of Flory's equation on page $\frac{1}{4}$:

Test	Per cent Bengcyl Peroxide	V2 Equil.	Mc
3a	0.5	.084	<u>4890</u> •525 - µ
3c	2.0	.206	1000 •588 - µ
31	8.0	•229	<u>838</u> •591 -4

(1)

D. W. Scott of the General Electric Company investigated the molecular weight distribution of a polydimethylsiloxane elastomer which he described as follows:

Fraction	Description	Per cent of Polymor	Average Molecular Weight
A	Elastic	17	2,800,000
В	Soft & Plastic	14	1,500,000
C	Soft & Plastic	18	610,000
D	Very Plastic	27	290,000
E	Viscous Liquid	15	Much less than 290,000

The elastomer used in this study, General Electric SE 76, is chemically identical with that used by Scott but the molecular weight distribution may be slightly different.

For calculation purposes, it was assumed that the average molecular weight of the polymer used in this

(1) Scott, D.W., J. Amer. Chem. Soc. <u>68</u>, 1877-1879 (1946)

study did not exceed 1,000,000. Since it was impossible that M_c , the molecular weight between cross-links, could exceed the linear molecular weight previous to crosslinking, the limiting value of $M_c = 1,000,000$ was applied to the above equation of Test 3a where M_c is given in terms of \mathcal{H} . This provided a limiting maximum value of $\mathcal{H} = 0.521$, and from this the limiting values of M_c were calculated for Tests 3c and 3i.

From Test 6 it was found possible to define a minimum possible value of M_c for sample 3c and hence for the other samples. Therefore, the correct value of M_c for the various samples of this test lies within the following range:

Test	Per cent Benzoyl Peroxide	Mc max.	Mc min.
3a	0.5	1,000,000	24,200
3c	2.0	14,900	6,600
3 i	8.0	12,000	5,400

Assuming that benzoyl peroxide causes cross-linking of silicone rubber by the formation of methylene groups, the decomposition of one molecule of benzoyl peroxide causes the formation of one cross-link between adjacent polymer molecules and hence two molecular chains between cross-links. Assuming such quantitative reaction, the minimum values of M_c for the various samples of this test would be:

3a
$$M_c = \frac{(100)(242.22)}{(0.2)(2)} = 24,200$$

3c $M_c = \frac{(100)(242.22)}{(2.0)(2)} = 6050$
31 $M_c = \frac{(100)(242.22)}{(8.0)(2)} = 1520$

The effectiveness of benzoyl peroxide, expressed as the ratio of the number of cross-links formed to the number that would be formed by quantitative reaction, is hence given by ratios of the above values of M_c ; values are tabulated below:

<u>Test</u>	<u>Per cent Benzoyl Peroxide</u>	Effectiveness
3a	0.5	2.4 - 100 %
3c	2.0	40.6 - 92 %
31	8.0	12.7 - 28 %

<u>Conclusions</u>: The swelling characteristics and the molecular weight between cross-links of silicone rubber are definitely functions of the amount of benzoyl peroxide used for vulcanization. The number of cross-links increases with the amount of benzoyl peroxide used, but the effectiveness of the latter reagent decreases when used at higher concentrations.

Use of Flory's equation for M_C of a polymer sample which is only slightly cross-linked (such as 3a) is theoretically inaccurate due to the large number of

9.

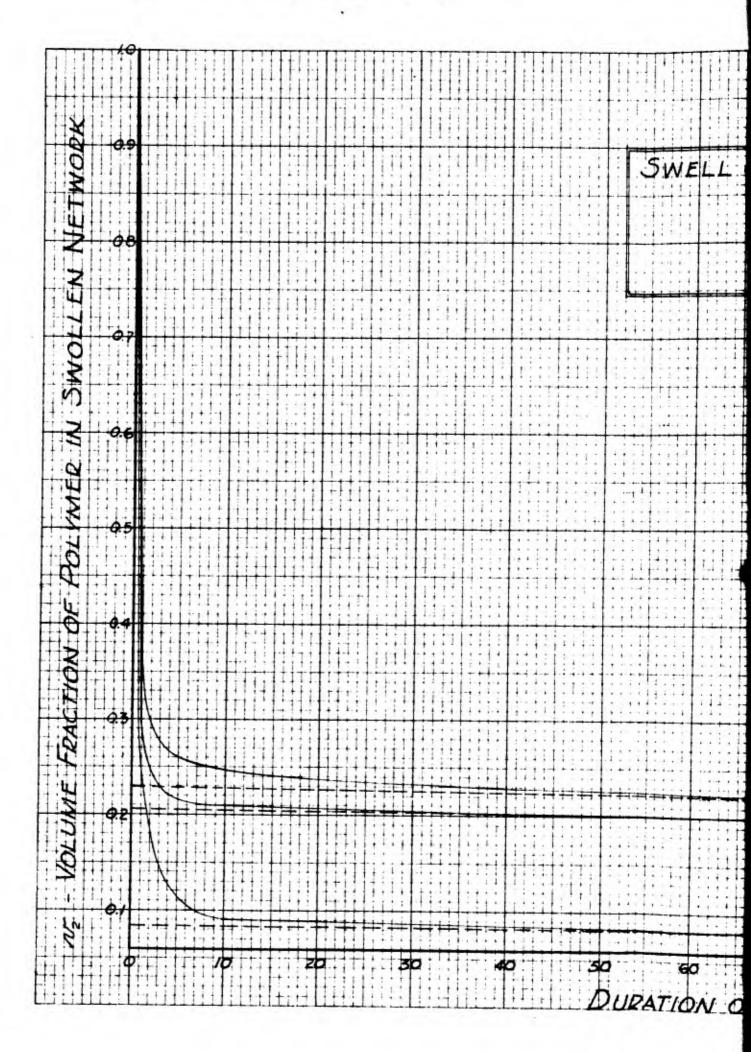
£

loose-ends and polymer molecules which are not a stable portion of the cross-linked network. Furthermore, where v_2 is extremely small, the denominator of the Flory equation becomes a minute difference between two relatively large numbers. Variation of v_2 by as little as 1 per cent can cause as much as 25 per cent variation of the calculated M₂.

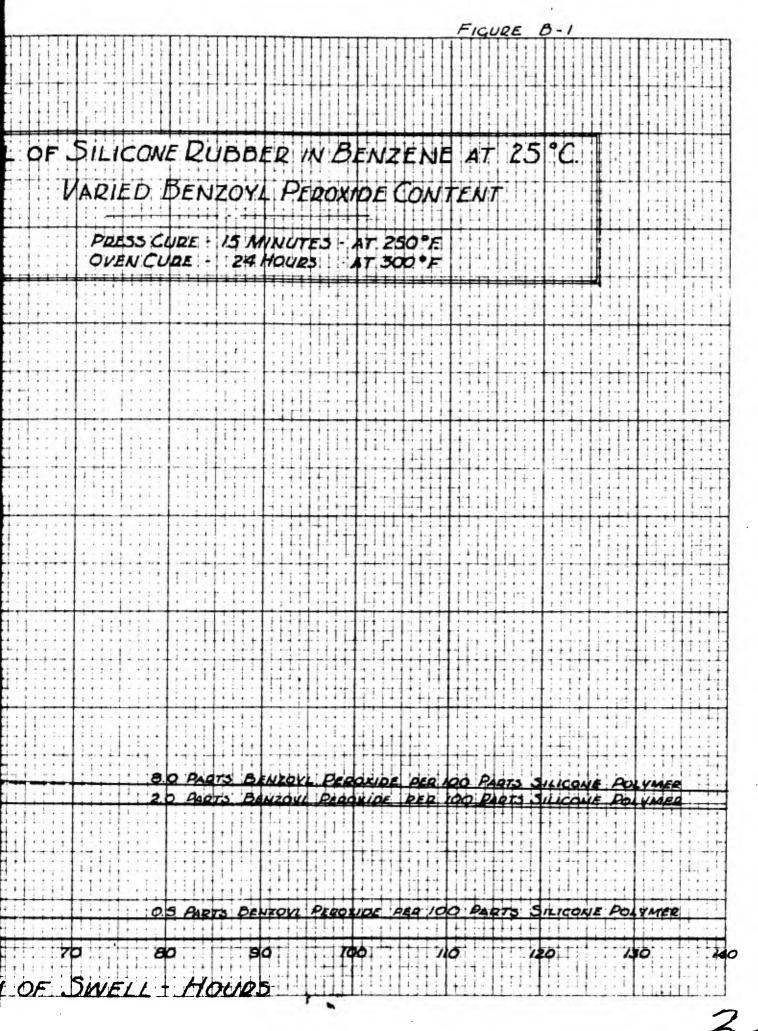
Volume measurements in this test are considered accurate to $\frac{1}{1}$ per cent; graphical calculations of v_2 at equilibrium are probably accurate to $\frac{1}{2}$ per cent.

ND 3400-10 DIETZGEN GRAPH PAPER 10 × 10 PER - NOH

ELGUAL DIETZGEN CD



and a second second



//

Test 4, Freet of Curing Conditions on Swelling

<u>Object</u>: The object of this test was to study various press and oven cure conditions of time and temperature and to determine the effect of these conditions on the swelling characteristics of silicone elastomer. <u>Procedure</u>: A large master-batch of polymer was prepared by milling together 2.0 grams of benzoyl peroxide with 100 grams of SE 76. This batch was separated into four portions and given four different press cures. Uniform oven cures were given to three of the samples, and two different oven cures were given to the other. These curing conditions are listed below and are graphed in Figures B-2 and B-3, pages 19 and 20.

<u>Slab</u>	Test	Press Time, Min.	Cure Temp. °F.	Oven Time:Hr.	Cure Temp, ^o F.
a	Ца	10	250	24	300
Ъ	4ъ	10	230	24	300
c	40	15	230	24	300
đ	4a = 3	c 15	250	24	300
d	Цө			1	300
đ	Цſ			48-hou at 480	cycle)°F.

The particular time-temperature cycle given to sample 4f was a standard factory-recommended oven cure.

The range of press and oven cure conditions in this test was designed to cover that normally encountered in industrial and laboratory vulcanizations of silicone rubber.

Loss of weight of the slabs during each curing operation was determined.

Samples were swelled in benzene at 25.0°F. and the rates of swell were determined during 5-minute to 145-hour periods. Loss of weight due to extraction during swelling was also determined.

The results of Test 3c were accepted for Test 4d since samples for that test were prepared according to conditions comparable to those desired for Test 4d; there may have been slight differences.

<u>Results</u>: Loss of weight of the slabs during press cure was independent of the curing conditions; loss of weight during the oven cure decreased slightly with increasing time and temperature of the previous press cure as shown by Table B-1, page 18.

As shown in Figure B-4, page 21, the rate of swell of silicone elastomer was only slightly affected by the press cure conditions. A considerable difference in the swelling of duplicate samples $4a_1$ and $4a_2$ was noted during the initial period of swelling, but the two samples exhibited the same rate of swell during the incremental period. For the graphical calculation of v₂, sample $4a_2$ was selected. Likewise, in Tests 4d, e, and f (Figure B-5, page 22) the rates of swell were only slightly different during the initial period (except for the anomalous behavior of sample $4f_2$), but there was a distinct difference in rate and magnitude of swell during the incremental poriod.

Equations for the rates of swell were as follows:

Variation of press cure

Test	Initial Period	Incremental Period
4a	.083 to 1.0 hours	7.0 to 145 hours
	$\ln v_2 =309 \ln(t) -1.309$	$\ln v_2 =0473 \ln(t) -1.507$
Цъ	.033 to 1.5 hours	5.0 to 175 hours
	$\ln v_2 =306 \ln(t) -1.204$	$\ln v_2 =037 \ln(t) -1.466$
4c	.083 to 1.5 hours	5.0 to 145 hours
	$\ln v_2 =226 \ln(t) -1.242$	$\ln v_2 =0611 \ln(t) -1.439$
4d	.033 to 1.5 hours	6.0 to 144 hours
	$\ln v_2 =310 \ln(t) -1.194$	ln v ₂ =0186 ln(t) -1.537

Variation of oven cure

4d	.083 to 1.5 hours	6.0 to 144 hours
	$\ln v_2 =310 \ln(t) -1.194$	$\ln v_2 =0186 \ln(t) -1.537$
Цe	.083 to 2.0 hours	5.0 to 145 hours
	$\ln v_2 =303 \ln(t) -1.213$	$\ln v_2 =0512 \ln(t) -1.516$
Цſ	.083 to 1.0 hours	5.0 to 145 hours
	$\ln v_2 =296 \ln(t) -1.22$	$\ln v_2 =056 \ln(t) -1.429$

In Figures B-6 and B-7, pages 23 and 24, are calculated the equilibrium values of v_2 according to R.L. Scott's definition which involves extrapolation of the incremental values to zero time. These values and the amount of extraction of polymer during the swelling period are listed below:

Test	Press Cure	<u>Oven Cure</u>	V2 Equil.	Per cent Extracted
4a .	10/250	24/300	.187	5.2
Цъ	10/230	11	. 205	6.0-6.5
40	15/230	88	.193	4.8
4a	15/250	11	.206	4.3
40	It	1/300	.180	6.5-6.8
Цſ	Ħ	cycle 480°F.	•195	5.5-7.2

The close comparison of the extreme tests, 4b and 4d, is contradictory to the trend of Tests 4b, c and a, which indicate a tendency toward decreasing v_2 with decreasing severity of press cure. It has been proven that benzoic acid, which is a product of the vulcanization reaction, causes depolymerization of silicone polymer. Since freedom of escape was restricted in the mold, the increased presence of benzoic acid may have counter-acted the increased cross-linking reaction during the longer duration and higher temperature of the press cure of 4d.

The amount of polymer extracted during the immersion in betweene (as well as the weight loss during oven cure) was inversely related to severity of the press cure, which indicated that longer time and higher temperature in this operation provided a more stable cross-linked network with fewer non-vulcanized and hence velatile or extractable polymer molecules.

The high-temperature oven cure of Test 4f probably caused some depolymerization of the network which resulted in a greater amount of swelling and of polymer extraction than was obtained with the more moderate cure of sample 4d. The slight oven cure of sample 4e led to a relatively high degree of swelling.

According to the previous and subsequent assumptions regarding possible maximum and minimum values of M_c and μ , values calculated from the Flory equation (page μ_{μ}) are listed below:

Test	Mo max.	Mc min.
4a	24,200	6970
4ъ	20,500	7600
4c	25,900	8840
4a	14,900	6560
40	38,000	10,900
Цſ	19,300	7830

Comparison of Tests 4d and 4e shows that the crosslinking reaction definitely continues during the 1- to 24-hour period of the oven cure at 300° F. <u>Conclusions</u>: The offects of moderate variation of press cure conditions (when followed by a good oven cure) on the swelling characteristics and on M_c of the silicone elastomer are slight. Longer press cures at higher temperatures lead to less volatilization during oven cure and less extraction during swelling of silicone rubber which has been vulcanized with 2 per cent of benzoyl peroxide.

The volatilization of polymer increases with time of oven cure at 300°F. and the molecular weight between cross-links decreases considerably during the same period. Since benzoyl peroxide is completely decomposed prior to this time, it is reasonable to speculate that volatilization of end fragments of the polymer chains may cause further condensation polymerization which would result in a more tightly cross-linked network.

Sample Preparation for Test L

,

Mester batch: Silicone polymer #81176, 100.0 grans

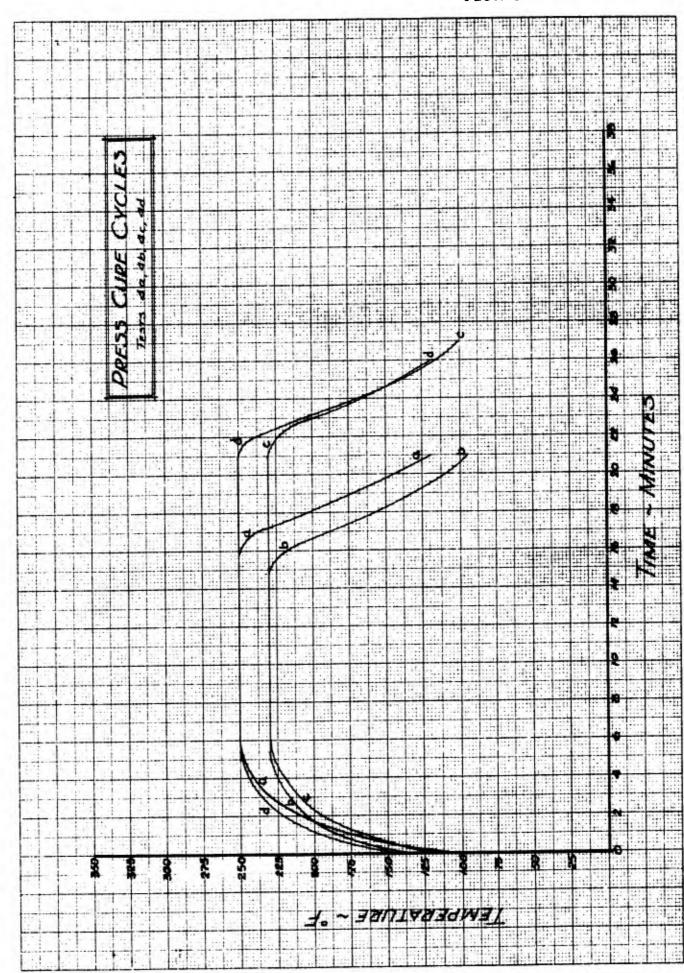
Benzoyl peroxide, 2.0 grams

Batch milled 15 minutes, separated into four portions for their respective cures.

							•	
	%Loss	8.19	8.37	7.88	• 723 5-66	6.65	9.25	
ems	Loss	91115.	·5418	• 5202	•723	-2742	•368h	
Weight,	Before After Loss	6.1046	5.9418	6.0825	13.061	3.3314	3.6106	
		6.6492	6.4.836	6.6027	300 12.784 12.061	3.6056	3.9790	
Temp.	Во	300	300	300	300	300	-	
Oven Cure Time. Tamp.	Hr	5	51	2 [†]	ħ	٦	48-hour	to 4800F
	Loss	0.63	0•149	0.73	16.0			
ht, gm.	After	21,.82	24.95	69·172	25.11			
Wels	Before After Loss	25.45	25.44 24.95 0.49	25.42	25.45			
Cure Temp.	·40	250		230		250	250	
Press Time,	Min.	10	10	15	15	15	15	
	Sample	6	م	U	Ð	e	٤.	

TABLE B-1

Note a - This data from Sample 3c



÷ .

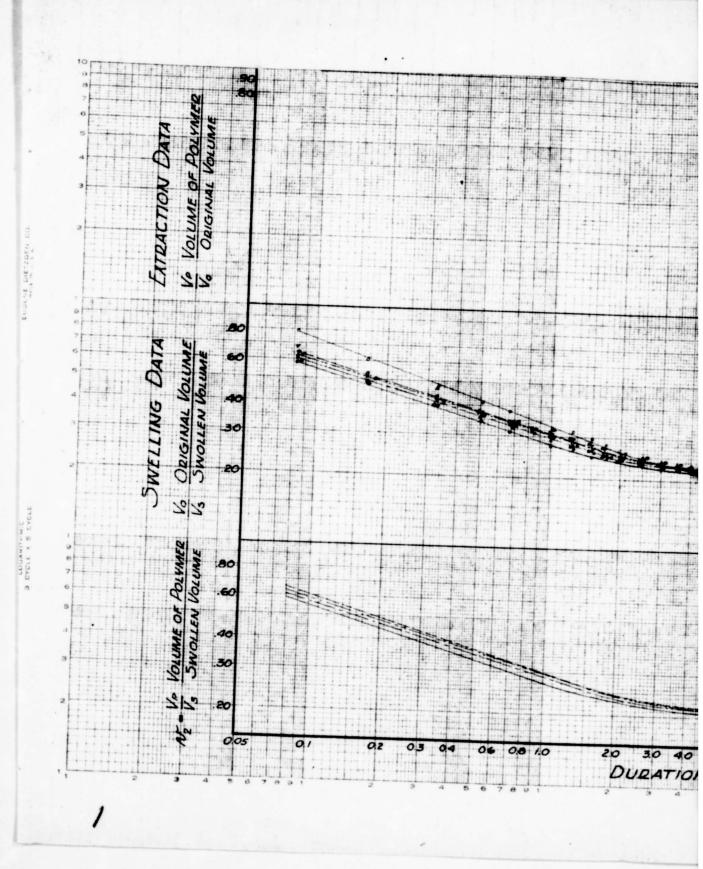
355T.14G REUFFEL & ESBER CO. Millimeters, 5 mm. lines accorted, cm. lines heavy vaue in y * a FIGURE B-2

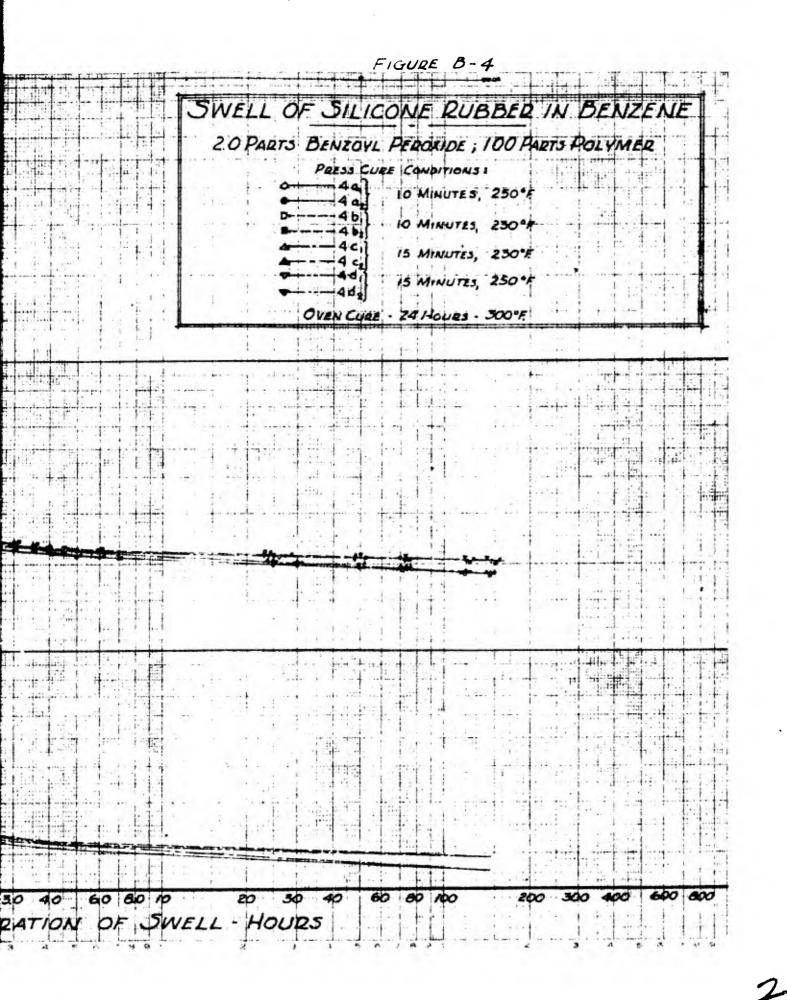
19

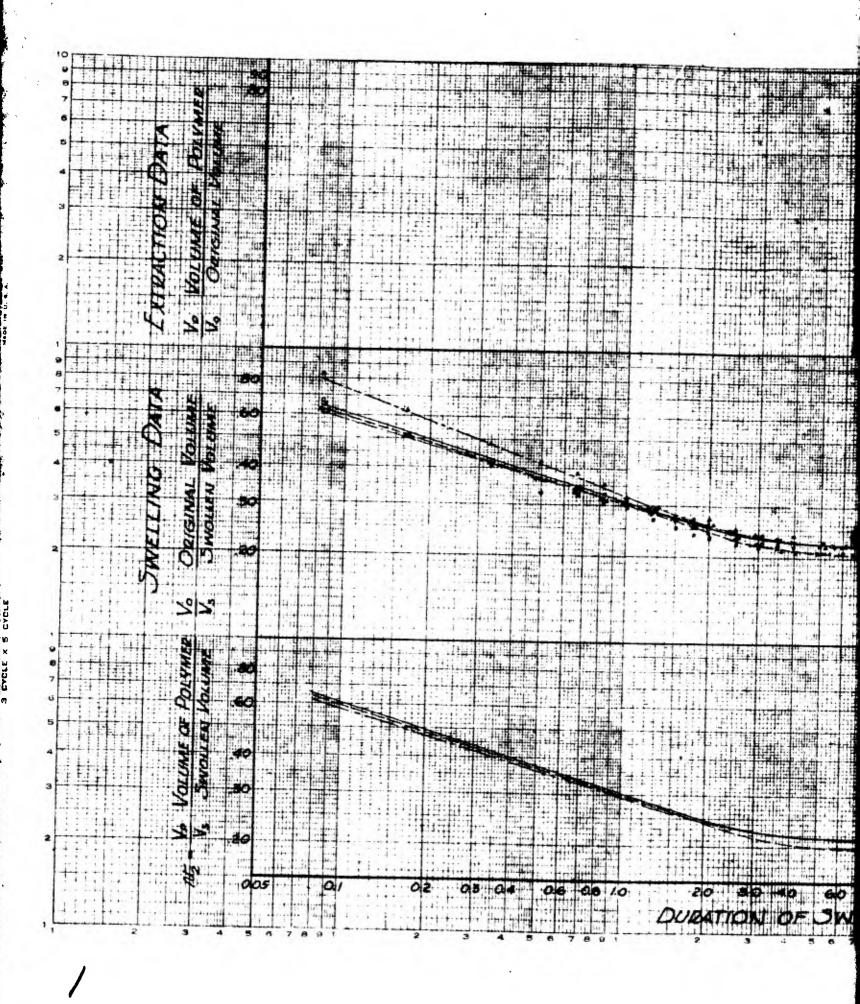
																											FI	θu	RE			>				
		1		T.				1		Ut:														H4		14				1	61	44			ii:	
7						1			i ii				ii.			1										ſ	11.2		1							
n i					i.																		in i		•											-
1	1.3				11					11			ПГ.,										HH:		-		111									1
	1.0												.H				1														1.11		•			
1		10												1																I.						
																											11.									
				1											i Li															Ħ	Тų.					
-					++++				F						14				Hit				Hi,					iir.								-
thi									117																											-
						+			111																											
				-																															E	
1		1			-											1																			F	1
														ł									in:						1011			. 11				
				1										1	11		tir:							1.44			1125						8		H	
		1		11		-																		12:1										1:11		-
		12							Hit						1				H								rfe.		1111				*		1	-
								-							Iλ									1												
	dei													111			11		1			1	8.0	i:E	Ξŧ	111	1.15	E.		1			8			
	172			1							114	1.	11			1		111			1.1		• 1				1. FL		÷.			••••	111		÷	1
	-3				D.	Н											N,				1			Щ.	11		111		θĦ	1			4			1.1
										ľ.													1	11.1				1.5	64			r	1			1
			17		N					111				Ľ.		fici	11	1 A				1					i T									
1.1					708.0				11	16	ΠĽ.								N								Π'n	1								1
			5				2	L			11			110			-			UT.				Π								i.1			h	ä
					NOV	121					1					Hif!	1411	HH		Ν.				Шł						H					E	i
							2			H.									1		H							ŧн,	51	H	H.				ł.	
		1.		11	4									h		ЦH.		111	1		N					1 H				Ш						Η
	1						2			in.			ri:			11	п				1						11.									
Ľ.	14			12				Ι		H		÷		1					HR;			V				Ξŧŧ	ii.					ł.,				-
	,	4-	-44		OPE		9			hij			14		17.							N								1						
									ų.	177				17											11		ШH.		1					111.1	.	
	-32		12.		2	11	14			i.						it.			rt.								20				ιH.					
101		-1	171	Ŧ				i.		1202	1							Hii		H			I	Π.	1		Η.		H. 1		1.1					
			. 11															1	11						iii.											
									4						1.4																					1
	-01								1								111	47				444,			. 117			44			1 1					
							i.	_		T.				IT.											14							17				
	1							11		1																				1						
		11	.41					ii.		111									4		1.111	114	I					11								1
								H.				i.f			t:															I						
						1				114				1.1				Ihi												Ħ			1.1	144	100	
-10			i.T				1						1.0.0											h												ij
	1.1																																			
								1.0											1.1.	111	11.1															E
								1					-	H	-								****			1111			X							
1111					17			14																				Щ	N							
1																														ŧ-						THE P
									i ii						5				ti !!									1.11		1					H	
						-										1			511																	
1	1.1	10				1		15				12		Hill	11	i:E		14		14年	111	11		dii.	1.1		17.	111	11	::H		1			131	1

355T-14G KEUFFEL A ESSER CO. Millimeters, 5 mm. lines accented, cm. lines heavy and 13 5 a

0. _______







•

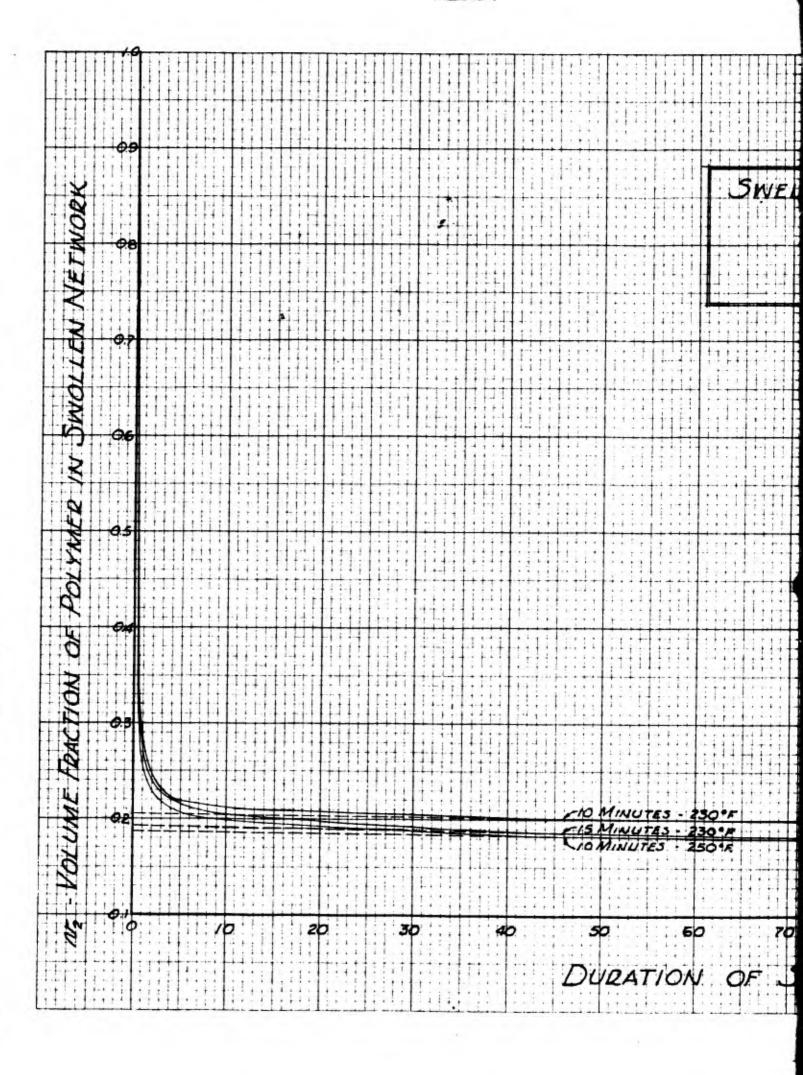
				FIGURE	8-5	אן כור ליין ייל הייכיץ, אורי חובר: ייק ב
	1550	ELL OF	SUICONE		2 IAS BENIZE	
		0 PARTS	BENZOVL P		DARTS POLYMER	
			PRESA GURE OVEN CUR	E Conorta		
		• 4 • 4	e. e. 2. 4. HOU		300*r	
			1			
0 40 6	0 80 4	0		60 60 10		
NOFO	WELL	- Houp	5	8 7 8 9 1	2	

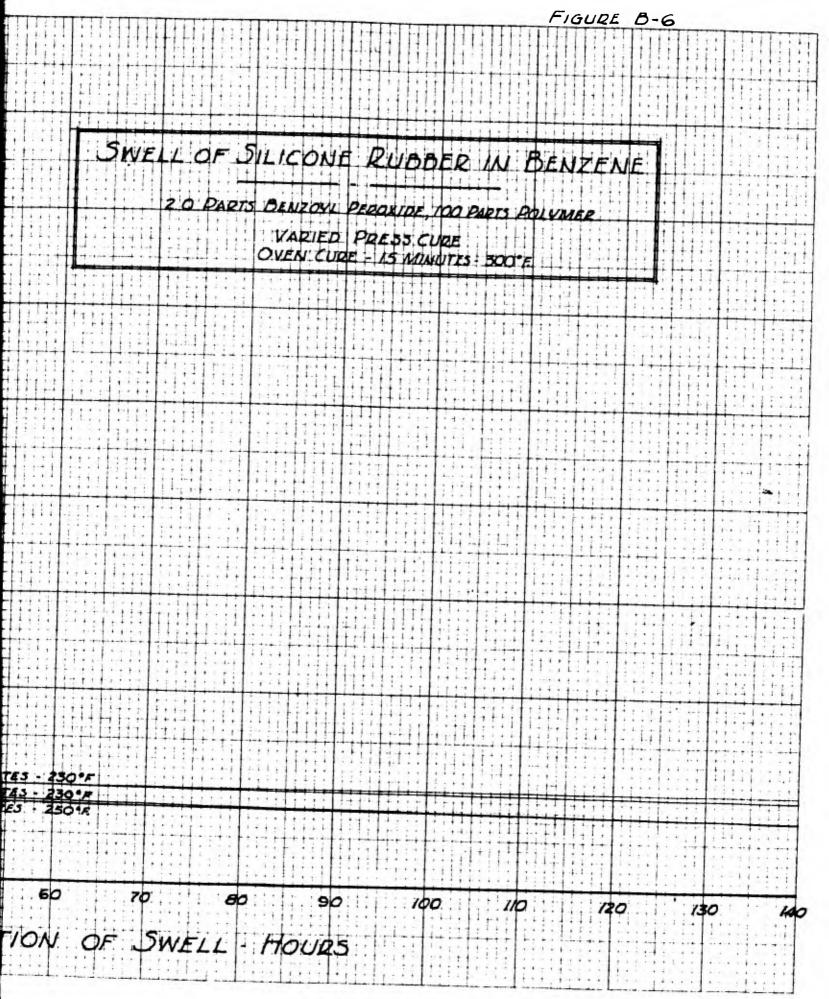
1617 - 18

ŧ

22

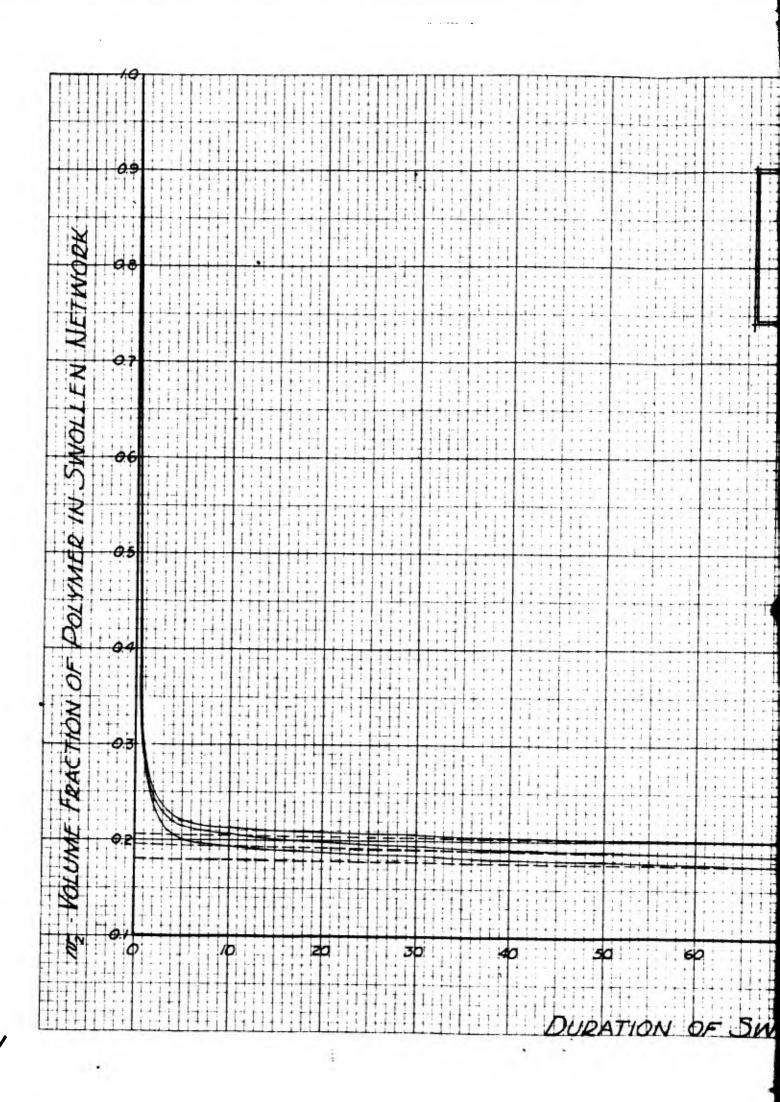
NO. 3400-10 DICTEGEN GRAPH PAPER 10 X 10 PER INCH

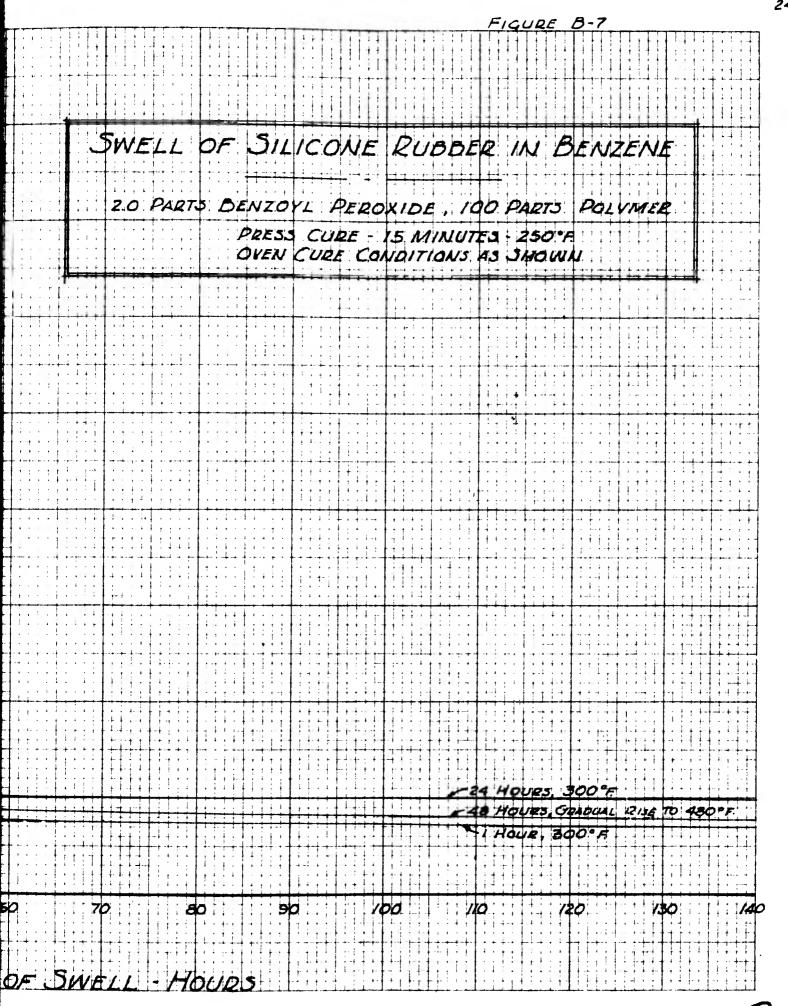




ND. 3400-10 DIETZGEN SPAPH FAPLR 10 X 10 PER INDH

EUGENE DIETZGEN CC.





Test 6, Swell of Silicone Elastomer in Various Solvents

<u>Object</u>: The purpose of this test was to investigate the degree and rate of swell of silicone elastomer in various solvents and to relate the results to current theories on the nature of the swelling of elastomers. <u>Procedure</u>: Samples of silicone elastomer were cut from slab 3c of Test 3, which had been compounded with 2 per cent of benzoyl peroxide, press cured for 15 minutes at 250°F. and oven cured for 24 hours at 300°F.

Wherever possible the ten solvents used in this test were distilled for high purity. Swelling tests at 25.0° C. were, in general, carried out over periods of 5 minutes to 145 hours, and the amount of polymer extraction was determined in each case. <u>Results</u>: The data of all tests except 6j are presented in Figures B-8 and B-9, pages 33 and 34. In Test 6j, swell in perfluoro(methylcyclohexane), the degree of swelling was less than 2 per cent and, considering the possibility of experimental error, the swell was considered approximately zero.

Rates of extraction were not determined for the various solvents; hence it was assumed that $\log({^{V_{\rm P}}/{\rm V_O}})$ was linear with log (t) as was found approximately accurate in earlier tests where ultimate extraction was less than 6 per cent. The ultimate extraction of

polymer at 145 hours varied slightly according to solvent in the range of 3 to 6 per cent. No correlation of the amount of extraction with solvent properties or with the degree of swelling was observed.

26.

The rates of swell of the silicone elastomer in the various solvents differed considerably as expected. In every case the duplicate tests provided comparable results. Equations for the rates of swell were calculated and are listed in Table B-2, page 35.

In Figures B-10 and B-11, pages 36 and 37, the rate of swell curves were extrapolated to zero time for determination of v_2 equilibrium. These values are listed below in order of decreasing v_2 and hence increasing swell:

Test	Solvent	v2 equil.
6j	Perfluoro(methyl- cyclohexane)	1.0
6h	Methyl ethyl ketone	•285
61	Benzene	•206
6g	d-Limonene	.183
6c	Xylene	.182
6a	Toluene	.170
6a	Cyclohexane	•148
60	n-Octane	.141
6ъ	n-Hexane	•138
6 f	Methylcyclohexane	.137

These values of v_2 were used in Flory's equation for M_c in terms of M and calculations are presented on page by

As all of the samples used in this test were from the same slab, the molecular weight between cross-links was constant throughout the test and μ was the variable. In Test 3, M₀ of sample 3c was calculated to be not greater than 14,900. This limit for M_c max. was applied to the equations of Test 6 and provided values of μ max, which are listed below on page 28.

By assuming that μ_1 from Huggins' osmotic pressure equation (page 71, Progress Report 10) was equivalent to M from Flory's equation, and by accepting D. W. Scott's value of $\mu_1 = 0.498$ for dimethylsiloxane polymer with methyl ethyl ketone, it was possible to calculate values of M_c and \mathcal{M} for each solvent of this test. However, this defined impossible negative values of μ and values of M_c which would require greater than 100 per cent reaction of the benzoyl peroxide. Hence, in order to define a lower limit for the range of M_c , it was assumed that M_{min} . = 0 for each of the solvents. This assumption defined the lower limit of M_{c} as 6560, which would be coexistent with # min. = 0 for polydimethylsiloxane and n-octane. By this definition, the following values of M min. were calculated for the other solvents:

Test	Solvent	Mmax.	
6h	Nethyl ethyl ketone		M _{min}
61	Benzene	•577	•531
6c		• 521	•436
	Xylene	•458	• 311
6a	Toluene	•439	•295
6g	d-Limonene	•416	
6a	Cyclohexane	·	•2 <i>2</i> 4
6 f		•400	•211
6ъ	Methylcyclohexane	• 333	.071
	n-Hexane	• 323	.068
6e	n-Octane	• 308	.000

P.

The above listing of solvents in order of decreasing \mathcal{M} does not agree with the previous tabulation of the solvents in order of decreasing v2. This difference points out a flaw in the basic method (but not in the theory) used by Geoffrey Gee for determining δ_2 , the solubility parameter of the polymer (pages 69-70, Progress Report 10).

According to Gee's method, Figure B-12, page 38, shows the relation of swell (here the inverse, since v_2 is plotted) to the solubility parameter of the solvent. Points are lettered according to the particular test number for each solvent. To draw a smooth curve representative of the points involved considerable approximation, and minimum v_2 (corresponding to maximum swell) defined \mathcal{S}_2 only vaguely.

Despite this poor representation of data, a value of $\delta_2 = 7.50$ was assumed and according to Gee's refinement, Figure B-13, page 39, was prepared with v₂ plotted versus $\sqrt{v_1}$ (δ_1 -7.50). The representation here was better than in Figure B-12 and was certainly as good as has been exhibited by Gee in his discussion of the method⁽²⁾. Nevertheless, even this supposed refinement does not define δ_2 accurately.

As an alternative to the above methods of Gee for determination of δ_2 , a more accurate and equally simple technique was devised which is presented in Figure B-14, page 40; as in Gee's method, the proposed improvement required only the knowledge of v_2 , V_1 and δ_1 . The present technique involved the use and the theory of Flory's equation and the qualitative application of various equations relating \mathcal{M} to $(\delta_1 - \delta_2)$. On page 44 were calculated approximations of M_c in terms of a uniform function, $(\frac{1}{2} - \mathcal{M})$. These values were plotted against δ_1 for the various solvents and, although there was some scatter, a sharp peak was defined at $\delta_1 = \delta_2$. This graph provided the value of $\delta_2 = 7.50 \neq .05$.

This approximation of Flory's equation would be absurd if used for calculation of M_c for some of the solvents since, for instance, with methyl ethyl ketone where M > 0.5, M_c would appear to be negative. Nevertheless, its use in the proposed method for determination of \mathcal{S}_2 is valid both in theory and in practice.

(2) Gee, G., "Thermodynamics of Rubber Solutions and Gels", in <u>Advances in Colloid Science</u>, Vol. II, Interscience Publishers, Inc., New York 1946, pp 145-195.

Using this value for the solubility parameter of polydimethylsiloxane, functions according to Gee's theories were calculated for the various solvents on page 44, $\frac{v_1(S_1 - S_2)^2}{RT}$. The range of possible values of \mathcal{M} for the various polymer-solvent pairs were plotted versus this function in Figure B-15, page 41, and provided a definite, but not linear, relation.

In order to ascertain whether the rate of swell of silicone elastomer might be a function of the rate of diffusion of solvent into the rubber, Figure B-16, page 42, was prepared in which the coefficient of the logarithmic rate of swell equation was plotted versus $2\sqrt{1}$ (where $2\sqrt{1}$ = viscosity, V₁ = molal volume) which was assumed to be related to the liquid diffusivity. No correlation was found to exist.

Further correlation was attempted in Figure B-17, page 43, where the slope of the rate of swell curve was plotted versus the equilibrium value of v_2 for all samples of Tests 3, 4 and 6. The resulting "path" was indicative that the rate of swell might be related to the free energy driving force which is, in turn, a complicated function of the changing v_2 and the constant \mathcal{A} and molal volumes. Of particular interest was the fact that the three samples of Test 3 (in which only M_c varied) provided an approximately linear relation of (-A₁ to v_2). In this graph the points are noted according to the test number.

<u>Conclusions</u>: The solubility parameter, or cohesive energy density, of polydimethylsiloxane is 7.50 $\frac{1}{2}$ 0.05 (cal./c.c.)².

The theories of Geoffrey Gee relating the polymersolvent interaction coefficient to the solvent molal volume and solubility parameter and the polymer solubility parameter have been at least qualitatively substantiated.

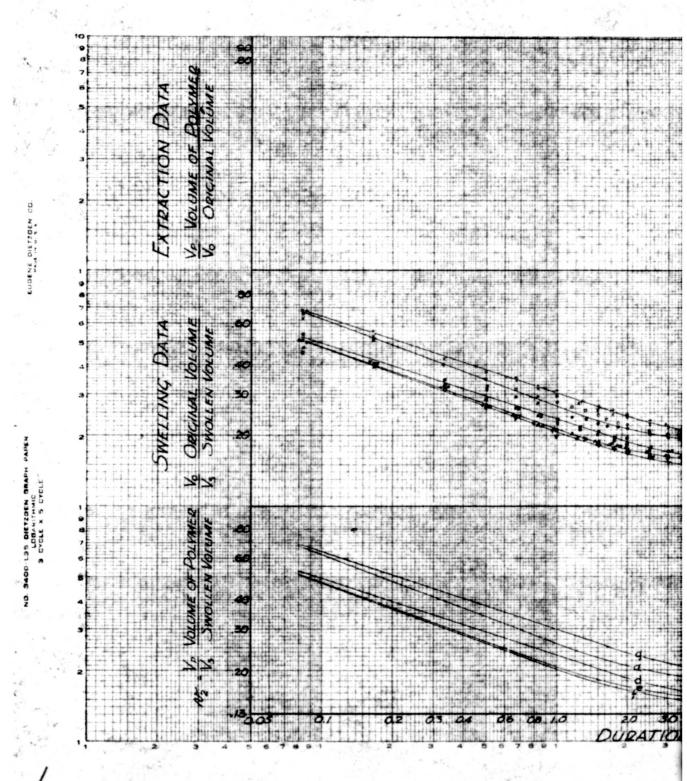
The polymer-solvent interaction coefficient, \mathcal{M} , for Flory's swelling equation does not correspond to the similar function in Huggins' osmotic pressure equation, \mathcal{M}_1 , for the polymer-solvent pair of polyaimethylsiloxene and methyl ethyl ketone.

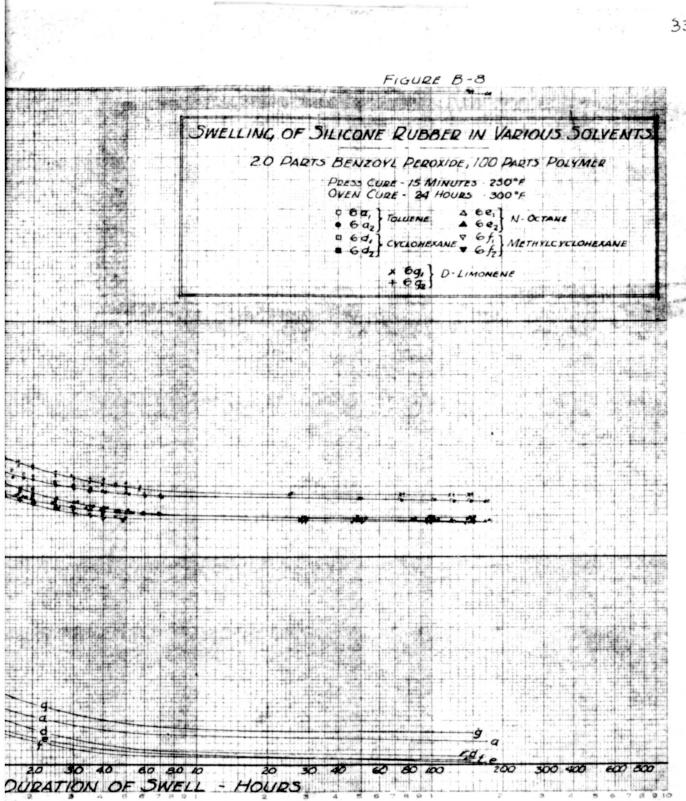
A method for the determination of polymer solubility parameter has been devised which in this test was as convenient to use and more accurate in results than the conventional methods of Gee. The proposed procedure might be applied to published results of the swelling of other elastomers in various solvents in order to test its general worth for providing sharp definition of S_2 .

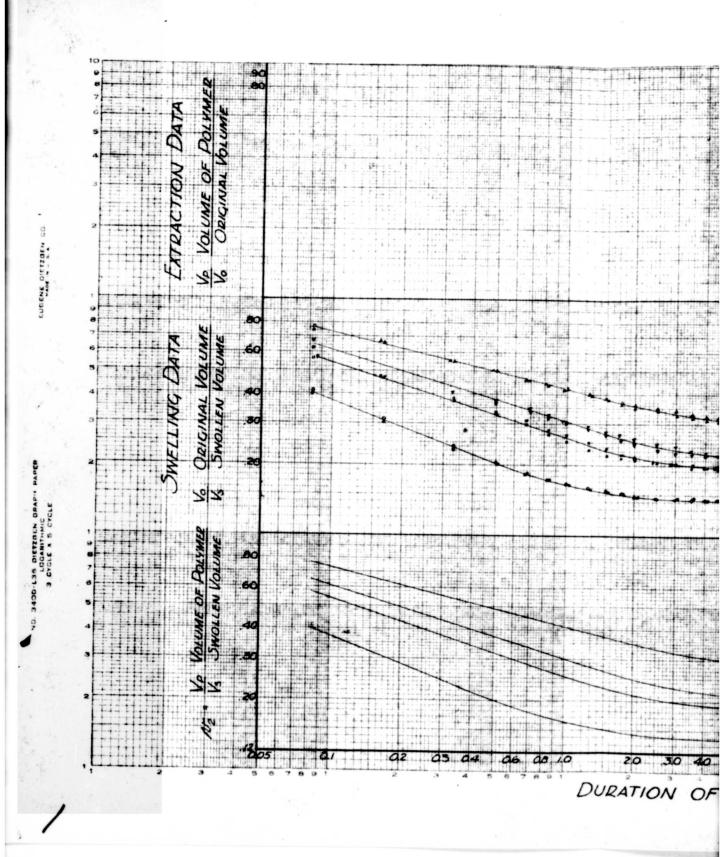
Values of M_C and \mathcal{M} have been defined within sufficiently narrow limits for comparative purposes; more accurate determination of the latter term is to be desired in order to correlate theory properly.

The rate of swell of silicone elastomer is probably dependent upon free energy factors rather than upon the rate of liquid diffusion into the elastomer. A The latter would probably become the rate-controlling factor, **Hereire**, if an extremely viscous liquid were to be used, such as a high molecular weight polymer.

The reasonable values of M_c and \mathcal{M} along with the aforementioned theoretical correlations add to the substantiation of P. J. Flory's equation for the swelling of polymers.







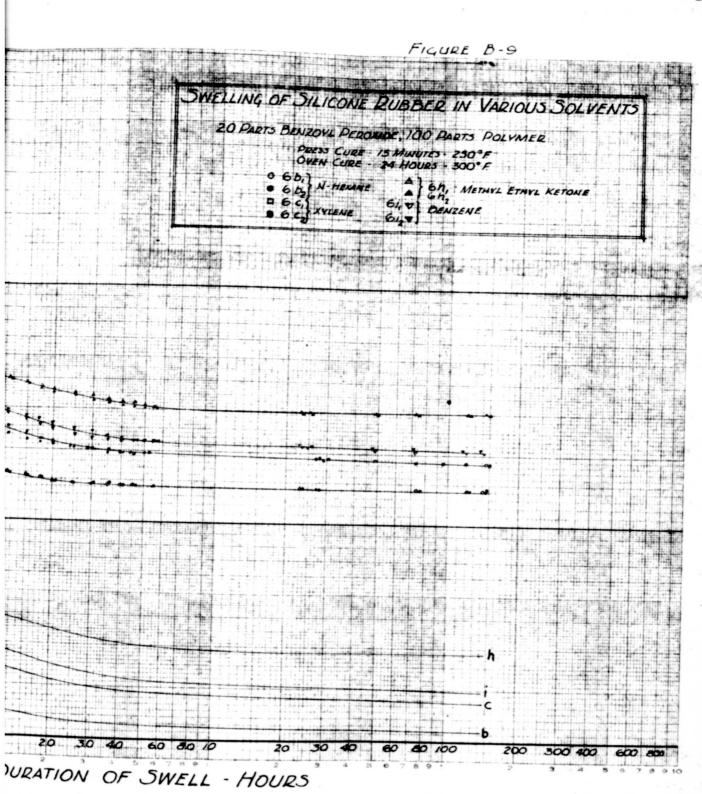


TABLE B-2

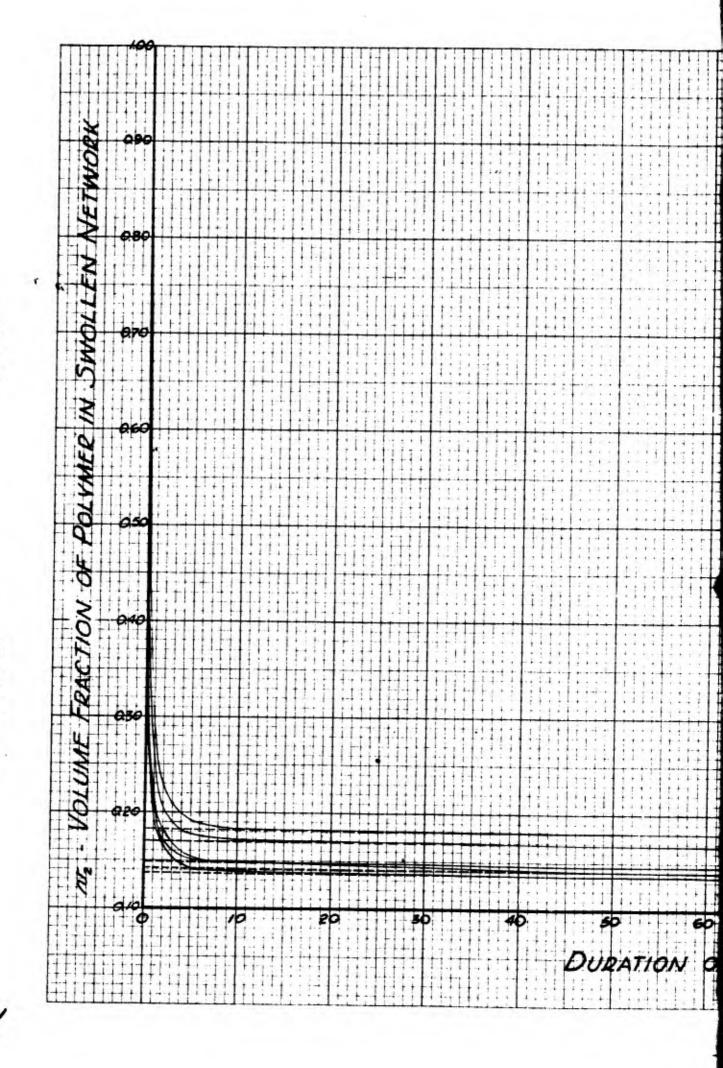
Rote of Gwell Equations for Test 6

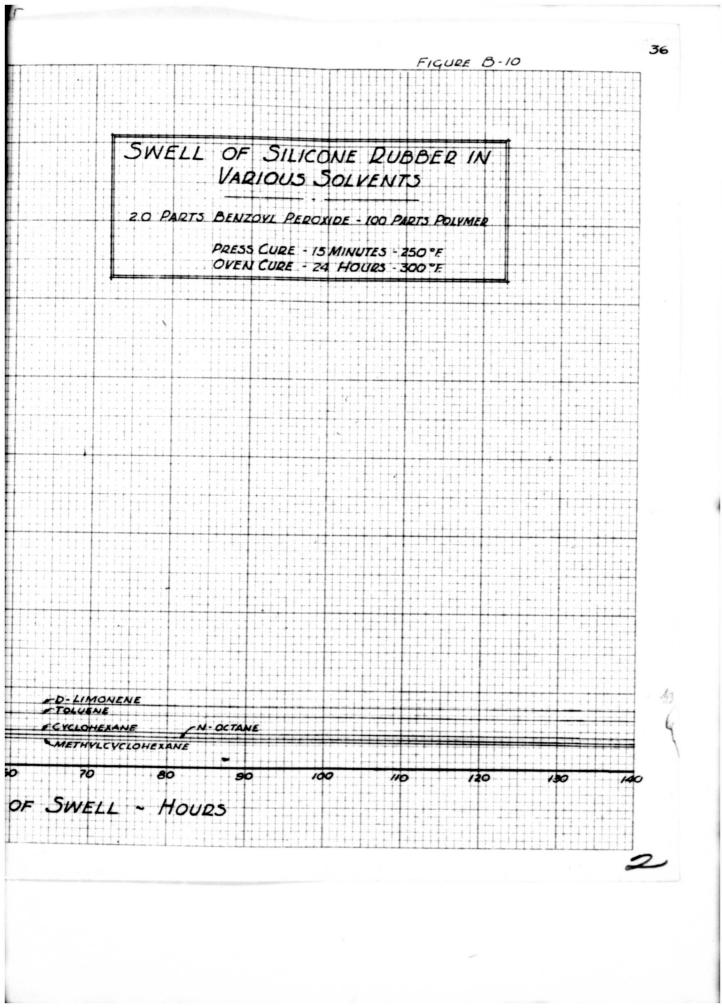
Test and Solvent

Solvent	Initial Period	Incremented Poulod
6a Toluene	.033 to 1 0 hours	7 to 175 hours
	ln v ₂ =380 ln(t) -1.347	$\ln v_2 =0222 \ln(t) -1.705$
6b n-Hexane	.083 to .40 hours	6 to 145 hours
	$\ln v_2 = -393 \ln(t) -1.833$	$\ln v_2 =0093 \ln(t) -1.971$
6c Xylene	.083 to 1.0 hours	10 to 145 hours
	$\ln v_2 =327 \ln(t) -1.370$	$\ln v_2 =0147 \ln(t) -1.218$
6d Cyclohexane	.083 to 1.5 hours	7.0 to 145 hours
•	$\ln v_2 =331 \ln(t) -1.581$	$\ln v_2 =0138 \ln(t) -1.890$
6e n-Octane	.083 to 1.0 hours	10 to 175 hours
	$\ln v_2 =359 \ln(t) -1.5702$	$\ln v_2 =133 \ln(t) - 1.339$
6f Methylcyclo-	.083 to 1.0 hours	7 to 145 hours
hexane	$\ln v_2 =372 \ln(t) -1.604$	$\ln v_2 =0119 \ln(t) -2.054$
6g d-Limonene	.083 to 1.5 hours	20 to 145 hours
	$\ln v_2 =378 \ln(t) -1.327$	$\ln v_2 =0085 \ln(t) -1.689$
6h Methyl ethyl	,083 to 1.5 hours	20 to 145 hours
ketone	$\ln v_2 =244 \ln(t)877$	$\ln v_2 = -1.255$
61 Benzene	.083 to 1.5 hours	6 to 144 hours
	$\ln v_2 =310 \ln(t) -1.194$	$\ln v_2 =0187 \ln(t) -1.537$

ND. 3400-10 DIETZGEN SHAPH PAPEN 10 X 10 PEH INDA

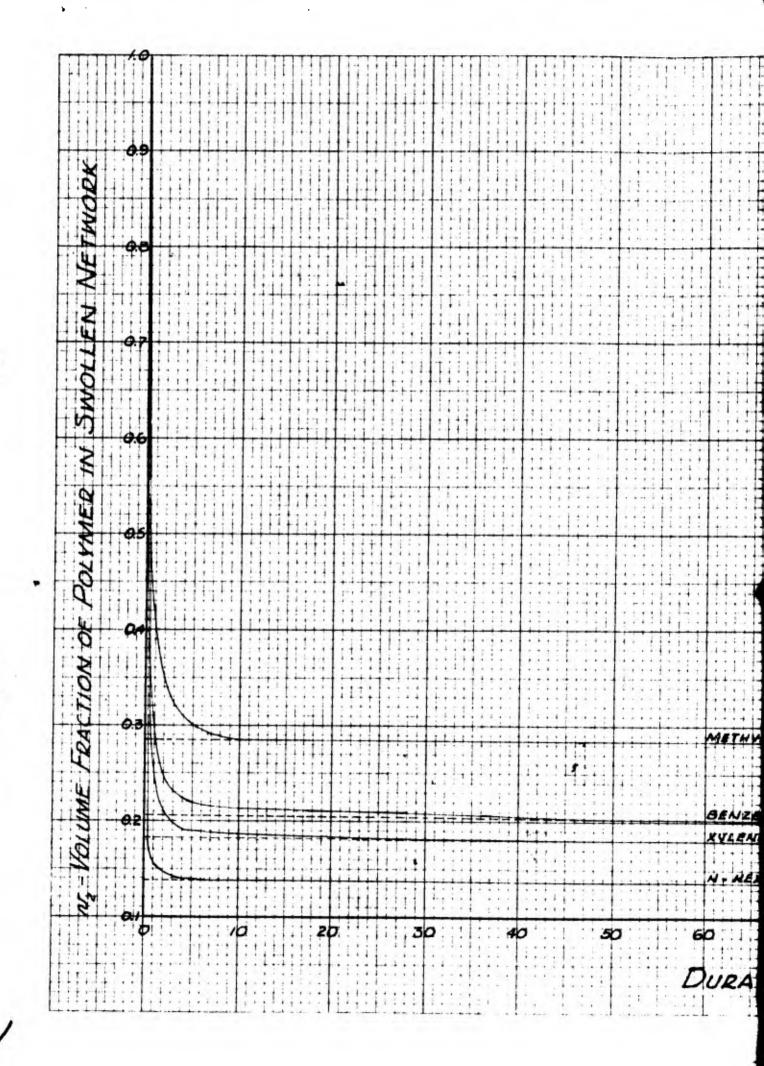
ELGENE DIETZGEN CU.



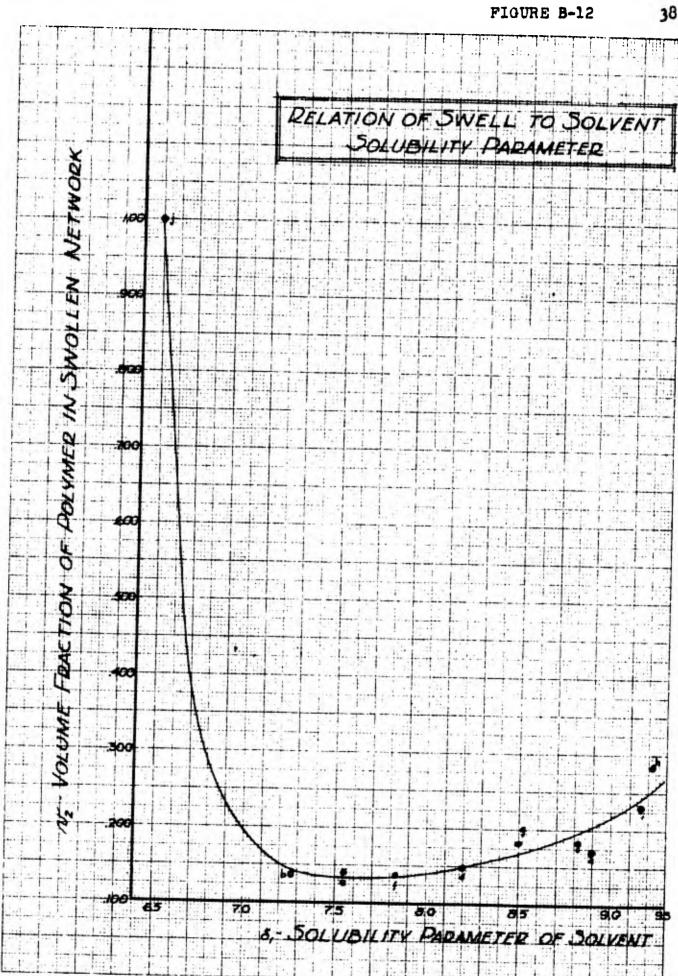


NU. 3400-10 DIETZGEN 38AUH PAPER 10 x 0 FLB - NEH

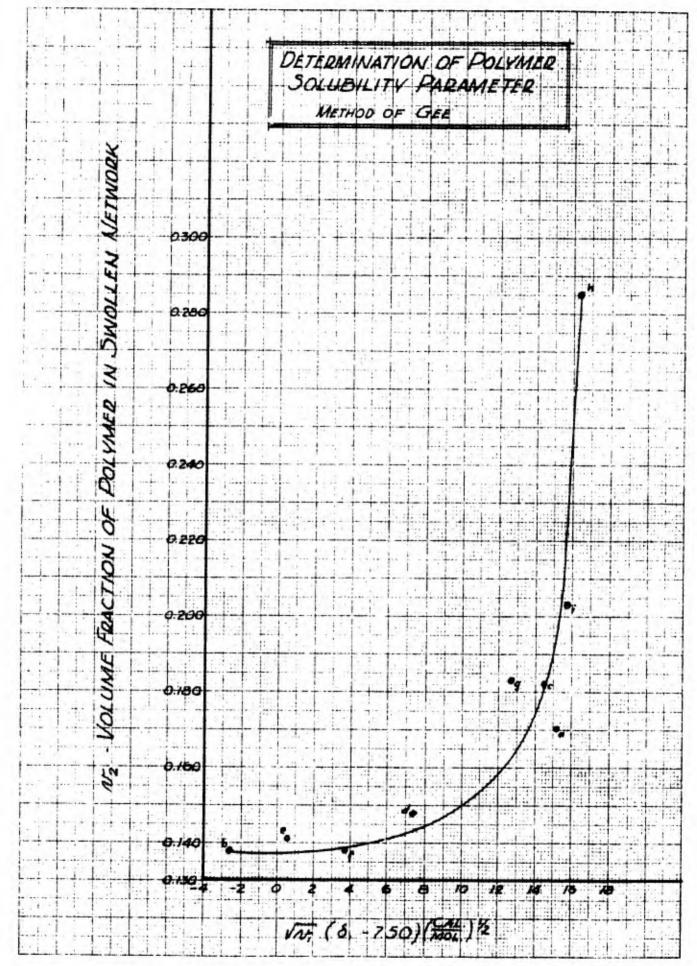
EUGENE DETZGEN CO.

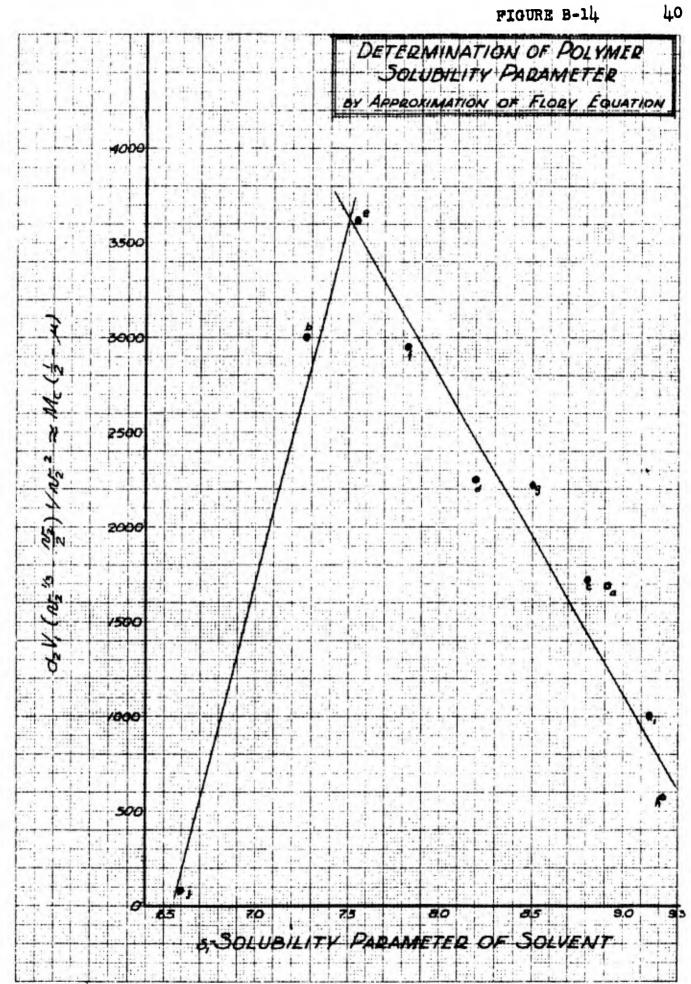


					T					F	igu	JRE	B	-//				
													Ti				1.1	
		1111				114	11	111	4	111-	11	4		!il				1
: 1	111	111											111	111		111		
11		5n	AFT	DE	50	100							1		11			44
				OF VAR					U	701	FK	1		11				11
it th			++++	KAK	100	2	p	VE	MI	3		+++		:4	1:1			14
								+ + + +		1141	and to b				ΗL	1:	:::	111
1.11			AKIS	BENZO	PED	XIDE	; 14		AP	P	PLY	MER	++	÷÷			+++	÷÷÷
		1111	1111	PRESS	CUR	F- /51	WAR	TES	-25	6 E		14		H.	11.			
111			1.1+1	OVEN	CURE	- 24	HO	UPS	30	\$ t		+++		++		+ :	11	111
					1;;;	dit:						111				$ \mathbf{i} $	11	11
			111	III II	1111				TI	IT		††		-	:++	1::	11	11
			1111		111	1			111			11.					11	
	i				;;;!	E	TH		11	ΠĽ	11	111				-11		Tit.
			1111	11111	1:11		411	111	+++	111	111	1:1.	111					I.t.
			1111			111	tti		1-1-1			111					11	
ali.		1.11			14	111	111		4				11	11		144		::::
		litt				141						1 H	::!	: 11		111	1.4	
		+			111			444		1.14	11	14.	11	14	111			
		<u> </u>		ΗH								1.1.1	1		11:	11	::1	
		4:44	444	1	+++++	4:44	1:11	14		1.	11			1	111		21	
• • • • • •				1111		ĽΗ		TH	11			H.		ili	14			
titt						+++		+++-	+++		++-	111		:::	1.1	44		11:
1111		11111	11:1			1111	+	H.	1 . • • •	11	iĦ			11	:::			
;; † ;;	::::::::	+	11	ttt	ttt	111	HH	+I+		111	1			11:				
	:HEE!			111		::::		1	11	114			:::			1		
THYL	ETHYL K	ETONE					+	11		:::	1:		++++	++		+++		
								IT.		1			111					1.1
				111		11-1-		Tt:	11	117	H-		+	-1-			÷ł,	
VZEN	£		1711					111		111	1		11.	11				
ENE			1 1						11			1-1	++-	-				-
		1.11	+		111					4		11	111					111
MEXAN		HHH			111	H		111		11	11	11		11			+++	
					1111			117				::	:::		11	1		
111.	70	80	x : ; ; [41:1	10	0		14	2:1:	111	12	0. :		1	0	111	: : 74
			1111				11.			1.1	1::		11:				:1:	1
ZAT!	ON C	DF J	WE	44 -	He	DUR	5		$\left \frac{1}{2} \right $				1	1.		11	1	
- 1 . 	· · · · · · · · · · · · · · · · · · ·		· · · · · · · · · · · · · · · · · · ·				* * *	111	-				::!:	÷	11	1.1	÷ ;	4



3597-140 KEUFEEL A ESSER CO Millincters, 5 mm. **lines accent** di. cm. lines hicavy. 4400. S. S. S. S.





sas7-14G KEUFEEL a Easer CO. Millimeters 5 mm. lines accented, cm. lines heavy.

		U COL	T								_	lg	170	3	-1:	2				-	+
		1.1.2					1			14			11	at	1	1					Ş
	1-1-	+-+ +	1	+	1 +	-				++	++					+	-	+-		-	0
一节的 二	1 h.		X	N	N		1	11	-	÷ŀ÷		-	1			4		d'e			
			1	Y		A				11			T		+	ť	+-	11-1	T	-	
1				1	1	T	X		1	1						1		1	1.19		Ś
					N	N						il r				11		1.1	To I		010
1.15.0.15		1			1	11			V							1			12	11	
				4		¥	$\boldsymbol{\lambda}$					ШĿ		1	i.	1					1
TON OF POLYMER - SOLVENT ON COFFICIENTS TO THERMO UNCTIONS TO THE POLYMER AND SOLVENT				-		1-		H		À			1							N.	
202				-			1 1	1	4		1				1			12		-	4
- SOLVENI TO THEE						-++		+	+-		- +		+	+	-	+	+			ir!	-
240				1	TT			1	-			4	1		-	!	÷	1.2		+	1
1- So 170									tt-	-	1	1	TT 1	-	-	+				-	11
272							11	T	ţ.,	15		1	11	1.	1	1	T.	1		1	0
CELATION OF PORYMER							T				hit.	i lei		1	T	1.	1	12.		R	000
3305				1.			4		1_		1	1			1	17			11		7.
8318				1	<u>-</u> [-		+	1			- 00	-				1.	1	1			
de Paryn Lefigirw 10k3 to Jolvent			-+-		-+-		-		4	-	4	-		1.	1.	in.	-	1	-		
4 2 2 3				-	- j -				-		-1-			-	-	11			. ink	4	
0.00								-	++-	-	-	+				-			-+	-	
CELATION C BACTION CO AMAE FUNICI AND					····	1.1	-		-		-	-							44	-	
286						11	-	T	tt		17	+		1	-	1	1		1	-	-
NTEPACTA DYNAME	1.1.4		1.19				0	1						17	1			14	T	15	ā
232			5,00				80		0.000			C			1		1.1		T	3	8
N 3 2							¥	1	90	1	1	6560		1		H.	10	1.	1		
						1.		1	1			9				1					-
		2011 (17) 2011 (17)			4	131	2	+-	M					41.5	1		31		11		
					1			15	1		+			1:-	+					1	-
				1		† +	1	1	1		ti:	+	++-	1		-		-		+	
								t:	1	-			1.17	11	-					-	
								1				1	tt		1						-
		1444			<u>[</u>].			1			1.		T		1.				1	QID	2
				411 1			-				-0			1	1					G	1
			-				-	1.			1		1	11	1			I	1	E	
		<u> </u>		-					1		4.			4					14		
	+ +		++				-11	-		++-			-1	+	-			-		-	4
				-			1	+		++			-11	V						-	-
		di t			1			1		H	1			+-				-	-	+	-
								1		1		t, t		-				-	t		1
								X			1			1			1.1			t	1
						1	•		N							Y	1			E.	
															-	,=			-	Ţ.	đ
	124	S 4.30		8			8			8			8		1	8				6	1
																				-	

•

																					F	IC	UR	B	в-	-10	ь		•		4
TT.	T		TT:	T			[]		1	-	1	111		-	1.	T	T	T			1		-	•	-	-		1 .	2		
17				+-	7	-	1			ļ	1.2	1	1	-	+-	+	1		Ţ				1					1.11	-		1
ng ng	-	-	-	-	+			1	-	-		-	+	-	ŧ	+		+	+	+	-						1				
1	10				1.			1	1	-			1.	1	4-	+	-+-	1	-			-	it.	-		-	-		92	_	1.1
1	Alat			-			i.	1	11	-	; ;		1	-	1	+	+	-	1		il.			-	÷			4			
4	4			T	1	F	1. 74 -	-	1 1 1 1	1.		-	1.,	1	1	T				-	1						1			-	1
		5 - E	-	t	1	+		+	3.	-		-	1	1	1	1	1	1		-+		-	-++	-	1				N-		- +
E	5	9	-		+-	1	-	1.	-	1-	1	1-		1	17	+		+						-		+	-	-			
MEL	-	ASSUMED		+	10		1-	1	1.		1	1	1	-		4.	+	1	-1		1			-			1		212		-
14	d	4		-	-	-				+	-		-	-	14	-			1						-		1				
DE LATE DE DIFFUSIVIT			in a	1		-	1	t		T	1.			T	+	-		-			44		•••		_			1.			
FFUSH	-	10	:+	+	+-	+-	1	+	1	1	÷	-	1	+		+	-	+	11				T	-				i.	NO	t	
				-	-		1	1	12	1	-	1.		-		1		-								-	1	145	-	<u>.</u>	
		INIT		T	1		1-	1	T	1	1	1			-			-					1				1	1	8	-	
NO		FULS			· ····	-	1	-	1	-	_		- 10	+.			-			-	+	-		-	4		1.	i.	-		EC.
WITH LIQUID	And Contract Manual Deputy In An	Unun Dreuwwirw				+-	1			•	-			1	1	-+	1	-										1		tor	~
J.H.	+	000	-	-		+	1	+	1	+-	1	+-		-	-	-						-	-	_			-	1	16	2	NU
Dac		110		-		1		-	-			-		+	1									-			-	-	-	<u>.</u>	1
ŭ				-		-		1	+	+	1	+	1	-	-							-		1 8		-	1	1	4		5
				-				-	1	-	+	T	-	-	1			<u>1</u> .		1		1		1	-		1	1	t		1
-+-				-+		-	1	-	Ť	-			1	1		-														1	
		111		+		+	+	+		-	-	\$		+	÷	-				1	+-						-		12		-
		1		-	-	1		-	1		1	1	1	-+	-1							1	-	-	-		1-	11	+		-
	-	1.	1.				++		-			+	-			-					-	i	-		Ľ	1	1		9		-
		1		-		+	-1		-				++							-	1	-		-	+	1.	-	1	-	+	+-
	-+			-	-	1	+	1	1		te de la	+		Ī						i				1.		1		1	4	4	
-1-				-	- +-	+	1	+		-	+		1	+					-	i.	-	1	1	T	-	1	-	1.	-	5	
	• {	Ţ.		-		1	-				1	-[-	-				0		-		i-	1	+	-	-	-		-	+-	-
						+	1	-		-			-			-			1										1		1
	- 1	-					-	0.0			1		0.35		1				020					0.25		1			2020	-	+
	00	1	-			1	+	1	~		~	,	- F				51		1.	74	4	1	R	12	-	77	-	4	12		

.

3537.140 KUFFEL A FSER CC. Millinnetes, 5 mm. lines acentral, cm. lines hered

1			T.		1		1	ł		-			+			-	1			-						1.15		8	-		-	
-											1.		+-					1				I	+9 e			1	.	1	1			
1		T	1	110		-						1		1				1	1	1	.								-			
++		JWE		5-3				1					1		1		1.1												–			
4	MEL	2	T									-		t				4				1		-	1			In				
	5	n al						1			-	1	1	1	+			1							1.	-		520				
T	OF	HIG					-					+	-	1																-		1.
	DATE	8	-						+				1			-		-			-	E.T.										
	a	8																										11		-	4	
	ð	-	-			1	F						1		-			-				1.1					1					
1	CORRELATION OF	WITH THE LIFGREE OF EQUILIBOUND JWELL				11	1		1							-	1	1										020		COULIBRIUM		
1	147	1	-									-	1					**										10.11		80		
	ZZE.	Ĩ	-							•			1		0 0		1		+			-		1						171		
	07	1							3		-		-						4											50		1
-		X	-					-				-			-	1								111								
													11	3										1				5		12.		
																												510				
										0					-					-												Ţ
Į.					i m:																				••••							
						ī						11					-	1														
							-1.																	10								-
											11					in .		1 142											n			
				1												H	i.														i i i	
						1	5					- 10																900				
						•••						1 0.35	111				200					0										
					5	*	74	A		4.2	4	7	2/	7. N)	C 7	7.7	75 44		4	4	7	10	70		1	1						
												-	-			1		>	II.										1			

399T-14G KEUFFEL & 685EF CO. Millimeters, 6 mm. lines accented, em. lines heavy FIGURE B-17 43

CALCULATIONS OF THE MOL

Tes	t Solvent	Molal Vol.		v ₂	¥ 2	√ ≜ <u>▼</u> 2	1 -• 2	ln(1 -y)	ln(1-y)
3 a	Benzene	89.40	.084	.438	.042	. 39 6	.916	0877	0037
3 c	Benzene		.206	.591	.103		•794	2307	025
31	Benzene		.229	.611	.1149	5 .4965		2601	031
4a	Benzene	89.40	.187	•572	.0935	5 .4785	.813	2070	020
4ъ	Benzene		.205	•590	.1025	.4875	•795	2294	024
Цc	Benzene		.193	• 57 8	.0965	.4815	.807	2144	021
Цđ	Benzene		.206	•591	.103	.488	.794	23 07	025
4e	Bensene		.180	•565	.090	.475	.820	1984	018
4 r	Benzene		.195	•580	•0975	.4825	.805	2169	022
6a	Toluene	106.85	.170	•554	•085	.469	.830	1863	016
6ъ	n-Hexane	131.60	.138	.516	.069	. ևև7	.862	1485	010
6c	Xylenà	123.46	.182	.567	.091	.476	.818	2009	÷.019
6 d	Cyclohexane	108.74	.148	•529	.074	.455	.852	1602	012
6•	n-Octane	163.52	.141	.521	.0705	.4505	.859	1520	011
6f	Methyl- cyclohexane	128.33	.137	.515	.0685	.44 6 5	.863	1473	010
6g	d-Limonene	162.2	.183	•568	.091	.477	.817	202	019
6h	Methyl ethyl ketone	90.16	.2 85	• 658	.1425	.5155	.715	33 55	050
61	Bensene	89.40	.206	.591	.103	.488	•794	23 07	025
6j (1	Perfluoro- methylcyclohexane)	173.3	1.00	1.00	0.50	0.50	0.00	- #*	

by approximate Flory equation $M_c = \frac{0.976 \nabla_1(\nabla_2 - \nabla_2)}{\nabla_2(\frac{1}{2} - M)}$

CONFFICIENTS

1 -• 2	ln(1-y)	ln(1-v) ≠v2	*2 *2	M * c approx	M _C **	M _C max.	M _c min.	M max.	M min.	δ,	δ,- 7.50	<u>V(5-7.50</u>) ² RT
.916	0877	0037	.0071	4890	4890	10 ⁶	24,200	.521	.436			
•794	2307	025	.0425	1000	.525 1000	14,900	6560					
.771	2601	031	•05 2 4	838	•588-4 838 •591-4	12,000	5400					
.813	2070	020	.0350	1210	<u>1210</u> س-571	24,200	8970	.521	.436			
•795	2294	024	.0420	1025	<u>1025</u> .571-#	20,500	7600					
.807	2144	021	.0372	1140	1140	25,900	8840					
•794	2307	025	.0425	1000	•565-4 1000 •500-44	14,900	6560					
.820	1984	018	.0324	1295	<u>1295</u> .	38,0 00	10,900					
.805	2169	022	•0380	1120	•555-4 <u>1120</u> •579-4	19,300	7830					
.830	1863	016	.0289	1695	<u>1695</u> •553 	14,900	65 6 0	.439	.295	8.91	1.41	• 359
.862	1485	010	•0190	3000	<u>3000</u> •525			• 323	.068	7.27	23	.0012
818	20 09	019	.0331	1728	1728 574-4			.1 78	. 311	8.82	1.32	. 364
.852	1602	012	.0219	2210	2210			.400	.211	8.20	0070	.090
.859	1520	011	•0199	3625	<u>3625</u> •552+4			. 30 8	.000	7.55	0.05	.0007
.863	1473	010	.0188	2960	<u>2960</u> .532-M			.933	.071	7.83	0.33	.0023
.817	202	019	.0335	2250	<u>2250</u> .567-4			.416	.224	8.5	1.0	•27
715	33 55	050	.0813	558	<u>558</u> .61644			•577	•531	9.22	1.72	.451
794	2307	025	•0425		1000			.521	.436	9.16	1.66	.415
0.00	- 04			1000 1000 85 1000	•588-14			not def	lined	6.5	-1.0	.29

 $= \frac{0.976 \, v_1(v_2^2, \frac{v_2}{2})}{\frac{v_2}{v_2} \, (\frac{1}{2} - \mu)}$

r

** by exact Flory equation $M_{c} = \frac{0.976 v_1(s_2^{\prime\prime} - v_2)}{\ln (1^{\circ}v_2) \neq v_2 \neq \mathcal{M} v_2^{\prime\prime}}$

de la

C. DISCUSSION

Without delwing into extensive theoretical assumptions, it has been possible to calculate within a fairly close range possible values of M_c and of \mathcal{M} for polydimethylsilexane swelled in various solvents. In addition, the effects of various conditions of preparation, i.e. benzoyl peroxide concentration and press and oven cure temperatures, have been discerned in their relation to the number of cross-links of vulcanized silicone elastomer.

Further work will be directed toward both theoretical and very practical considerations of this problem. From homogeneous molecular weight fractions of silicone elastomer it will be possible to define more closely the possible values of Mc max. and Mc min., and hence the values of // will be defined within narrower limits for the various polymer-solvent pairs. By treating swelling measurements in the proper manner, it will be possible to study rigorously the degree to which pigments reinforce silicone rubber and it may be possible to discern the mechanism of this reinforcement. A proliminary experimont has shown that swelling is reduced by pigments by a dilution effect and by a reinforcing action. Further studies will attempt to show the effect of various volume loadings of pigment on these swelling characteristics and various pigment types and sizes will be studied.

Of particular interest toward both theoretical and practical considerations will be a study of samples compounded with DuPont GS199S pigment to which no benzoyl peroxide has been added. This study should provide knowledge of the number of cross-links effected by the coated pigment and it should similarly permit differentiation of the reinforcing effect due to particle size and that due to the organic surface coating of the pigment.

Swelling measurements with the apparatus used in this study and with the unique procedure for graphical calculations should provide a valuable tool for future study of the physical as well as the thermodynamic properties of elastomeric polymers.

R. L. Hauser, Chemist

a. F. Elime

A. C. Glime Rubber Technologist

D. R. Mauly D. R. Manley

Laboratory Technician

REPORT SUBMITTED BY:

F. L. KILDOURDO, Jr.,

ç.....

t

Director of Research.

J. Stubblebine,

Director of Development.

Caraller C. A. Walker, Consultant Chemical Engineer

APPROVED BY:

m. Doete

C. M. Doede, Vice President

SUPPLEMENTARY

INFORMATION

TAB NO. 70-3	
--------------	--

1 February 1970

· · · · · · · · · · · · · · · · · · ·	Substanting Sections of the section		and an and the second s
IDENTIFICATION	FORMER STATEMENT	NEW STATEMENT	AUTHO RITY
AD-494 888 Connecticut Hard Rubber Co., New Haven. Progress rept. no. 11, 5 Dec 51-4 Jun 52. 5 Jun 52 Contract DA-44-109- QM-64	DDC users only.	No limitation	USAMC ltr, 19 Jun 69