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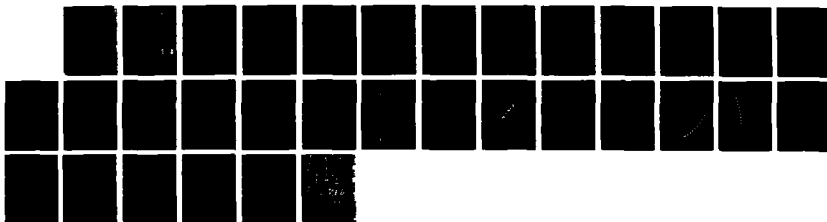
ELASTIC BEHAVIOR OF A RUBBER LAYER BONDED BETWEEN TWO
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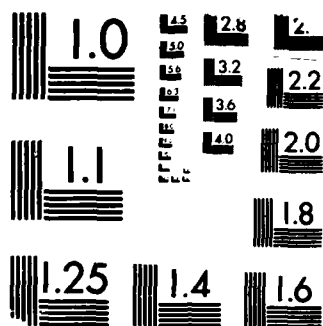
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ELASTIC BEHAVIOR OF A RUBBER LAYER
BONDED BETWEEN TWO RIGID SPHERES

by

A. N. Gent and Y.-C. Hwang

Institute of Polymer Science
The University of Akron
Akron, Ohio 44325

May, 1988

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20. ABSTRACT (Continue on reverse side if necessary and identify by block number) → Finite element methods (FEM) have been employed to calculate the stresses set up in a thin rubber layer, bonded between two rigid spheres, when small tensile or compressive deflections are imposed. Values of stiffness have been calculated for various spacings of the spheres, i.e.,		

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20. ABSTRACT (continued)

for various thicknesses of the rubber layer. They are in good agreement with earlier experimental measurements of compression stiffness and with the predictions of an approximate theoretical treatment (1). However, they are strongly affected by small departures of the rubber layer from complete incompressibility. The highest dilatant stress (in tension) is found to be set up on the central axis, near the bonded surfaces, where internal failure has been observed to occur in similar bonded layers (2). For moderately thick layers, the axial tensile stress in the center of the layer is substantially higher than the lateral stresses. This feature suggests that the initiation of failure in this location may not obey the same criterion as for an isotropic stress field, and that a crack, once formed here, will propagate as a tear across the axis of symmetry.

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Introduction

The elastic behavior of a rubber layer sandwiched between, and bonded to, two rigid spherical surfaces is of interest for at least two reasons. Structures of this type are used as flexible mounts and cushioning devices, and design engineers need to be able to estimate the stiffness and the stresses set up within them. And they constitute a model of particle-filled composites, in which spherical particles are dispersed in a softer elastic medium. In this case a knowledge of the stiffness and stress distribution may provide insights into the phenomenon of reinforcement.

Elastic behavior under small compressions (and, equivalently, under small tensile deformations) has been analyzed previously, using some rather drastic approximations (1). The rubber was assumed to be linearly-elastic and incompressible in bulk, and the compressive force was assumed to consist of two terms : one due to simple compression of the layer, and a second arising from a hydrostatic pressure $\underline{P}$ set up as a result of the restraints at the bonded surfaces, where $\underline{P}$ was a function only of the lateral distance of a point in the layer from the central axis of the system. We now present a more detailed and accurate study of the stresses and deformations set up by compression or extension of the layer, using finite element methods (FEM) and not invoking the condition of incompressibility. Instead, values of Poisson's ratio lying between 0.45 and 0.5 have been employed, covering the widest range likely to be encountered with rubber formulations. A typical rubber vulcanizate has a value of Young's

(tensile) modulus E between about 2 and about 10 MPa, and a much larger modulus K of bulk compression, of about 1.1 GPa (3). Thus, Poisson's ratio ν , given by

$$\nu = (1/2) - E/6K \quad (1)$$

ranges from about 0.4985 for relatively stiff compounds to about 0.4997 for relatively soft ones.

In addition, the distribution of pressure throughout the thickness of the rubber layer has been calculated, whereas, before, the approximate theory only yielded a uniform value.

These results have implications for the mode of fracture both of bonded rubber layers and of filled rubber and other particulate composites. Failure of rubber is known to take place where, and when, a triaxial tension (negative hydrostatic pressure) is set up that exceeds a critical value, given by about $5E/6$, where E is Young's modulus (4-6). Under these circumstances, any small spherical cavity that is present within the rubber will expand indefinitely, i.e., until the rubber around it reaches its maximum extensibility. Then the rubber will tear open to create a large internal crack. Thus, a crucial question for elastomeric composites is : Under what circumstances and at what locations is this critical state likely to be reached ?

A second question is : In what direction will the crack, once formed, tend to propagate ? If it grows across the sample, then it will result in rupture. If, on the other hand, it grows parallel to the axis of the sample, then it will not necessarily lead to rupture.

Some preliminary conclusions on these points are reached here on the basis of the calculated stress distributions.

2. Finite Element Analysis

A sketch of the model structure is shown in Figure 1. Stress distributions within the rubber layer were calculated using the ADINA code (7), assuming that the structure was cylindrically symmetrical, and that the rubber was linearly-elastic, isotropic, and nearly incompressible in bulk, with values of Poisson's ratio, ν , between 0.45 and 0.4999. The two spheres were made effectively rigid by giving them a value of Young's modulus of 10^9 times that of the rubber.

Ten equal elements were employed vertically, between the surface bonded to one of the spheres and the center line of the rubber layer. Ten elements of gradually increasing width were employed laterally, between the cylindrical axis and the free surface, as shown schematically in Figure 1. Values of stress were calculated by taking an average over four integration points of the eight-node axi-symmetric elements.

When an axial force F was applied to the spheres to compress or stretch the layer bonded between them, Figure 1, the FEM computations yielded corresponding values for the displacement $\hat{\delta}$, axial stress σ_z , radial stress, σ_r , tangential stress σ_t , and hydrostatic tension, $-P$, where

$$-P = (\sigma_z + \sigma_r + \sigma_t)/3 \quad (2)$$

These results are reported here for a wide range of layer thicknesses and for various values of Poisson's ratio, ν .

3. Results and discussion

(i) Stiffness of a bonded layer

Values of the computed compression or tensile stiffness for a bonded rubber layer are plotted in Figure 2 against the corresponding value of Poisson's ratio for the rubber. They are given in reduced form, as the dimensionless ratio, $F/\delta DE$, for selected values of the ratio h/D of the distance h between the spheres to the sphere diameter D .

As the results given in Figure 2 show, the stiffness of thin layers is extremely sensitive to the value of Poisson's ratio, even when it is quite close to 0.5, corresponding to complete incompressibility. For example, for a layer with a thickness ratio h/D of 0.01, the stiffness is reduced by nearly 70 percent in comparison with the result for an incompressible material, when $\nu = 0.45$. Even for a value of ν of 0.49, close to the incompressibility limit, the computed stiffness is still about 40 percent less than the value for a truly incompressible material. On the other hand, the stiffness of thicker layers is much less sensitive to small departures from complete incompressibility (Figure 2).

By extrapolation to $\nu = 0.5$, values of the stiffness of thin layers were obtained for comparison with those deduced previously from an approximate theoretical treatment for incompressible layers (1):

$$F/\delta DE = (\pi/8) \left[\text{Aln}[A/(A-1)] - 1 + (1/2A) + [1/(A-1)] \right] \quad (3)$$

where $A = 1 + (h/D)$. This comparison is made in Table 1 and Figure 3. The extrapolated results are seen to be in close agreement with the approximate theory over the entire range of rubber layer thickness. The maximum difference is about 6% , when the rubber layer thickness is relatively large, $h/D = 0.2$. Thus, the approximate theory is surprisingly successful in predicting the stiffness of thin incompressible rubber layers.

Because the approximate theory gave results in good agreement with experimentally-measured compression stiffnesses for bonded rubber layers (1), we can conclude that the numerical calculations are also in good agreement with experiment. This comparison is included in Figure 3.

(ii) Stress distribution within the layer

Values of the hydrostatic pressure P for compressed layers, or triaxial tension (negative hydrostatic pressure) for layers subjected to tensile loads, were computed by FEM, using Equation 2. Maximum values were found to be developed on the central axis. Results for the layer center are plotted in Figure 4 against the value of Poisson's ratio, ν , of the rubber. They are seen to be quite sensitive to small departures from complete incompressibility, especially for thin layers, as found before for the layer stiffness, Figure 2. (Note that logarithmic scales are employed in Figure 4, in view of the wide range of pressures encountered.) And, again, the results for thicker rubber layers were less sensitive to the exact value of Poisson's ratio.

By extrapolating to a value for ν of 0.5, corresponding to a completely incompressible rubber layer, results were obtained for direct comparison with the predictions of the approximate theoretical analysis (1), where the total pressure is assumed to be made up of two components, the first given by simple compression of an (unbonded) incompressible layer

$$P_1 = E\delta/3h \quad (4)$$

and the second arising from restraints at the bonded interfaces

$$P_2 = E\delta/4AD(A-1)^2 \quad (5)$$

where δ is the displacement, calculated from the total stiffness, Equation 3.

This comparison is made in Figure 5. Good agreement is seen to hold between values of pressure at the layer center, calculated from FEM and from the approximate theory.

Thus, the approximate theory is apparently able to predict the pressures set up in thin incompressible layers with surprising accuracy.

Up to this point, pressure were evaluated by FEM at the center of the elastic layer, for comparison with predictions of the simple approximate theory. But the FEM calculations revealed that, although maximum pressures were, indeed, developed on the central axis, they were not generally constant through the layer thickness. When the layer was extremely thin, the pressure was approximately uniform between the two spheres, Figure 6. But when the layer was thicker; for example, when $h/D = 0.1$; then the hydrostatic tension near the bonded interfaces was significantly larger than in the center of the layer, Figure 6. Thus, for thin layers, failure due to the action of a hydrostatic tension could occur at any point along the axis between the two spheres, whereas for thicker layers it is more likely to take place near the bonded surfaces. Experimental studies have shown that the first cavity appears in relatively thick layers near the bonded interfaces (2).

(iii) Anisotropy of stresses

FEM computations also revealed that the principal stresses set up along the center line were not strictly equal. They approached pure triaxiality for thin layers but for moderately thick ones the axial stress at the center of the layer was considerably larger than the radial and tangential stresses (which were roughly equal), Figure 7. However, near the bonded surfaces the stresses remained substantially equal, even for thick layers.

We conclude that failure by growth of a pre-existing cavity at a critical level of triaxial tension will occur first near the bonded surfaces for moderately thick layers, $h/D \approx 0.1$, and somewhat later, i.e., at a higher applied load, in the central region of the rubber layer, as observed experimentally (2). But the nature of the second fracture, and in particular the direction of tear propagation, is likely to be somewhat different because the stress field is not isotropic.

Conclusions

Stiffnesses calculated by FEM for thin incompressible rubber layers sandwiched between, and bonded to, two rigid spheres are in good agreement with a previously-derived approximate theory and with experimental measurements of compression stiffness. However, the calculated stiffnesses of thin layers are extremely sensitive to the value chosen for Poisson's ratio. Small departures from complete incompressibility bring about large reductions in stiffness.

The highest level of pressure in compression, or dilatant stress (triaxial tension) in tension, is developed at points near the bonded surfaces of the spheres for moderately thick rubber layers. This is the place at which an initial cavity appears when bonded rubber layers are subjected to tension (2).

Calculations of the radial and tangential stress show that the stresses in the center are strictly triaxial only for thin layers. For thicker layers, the axial tensile stress is substantially greater than the lateral stresses. This feature of the stress distribution has implications for the direction of tearing when an initial cavity forms at the center of the rubber layer. The tear will presumably run at right angles to the major tensile stress; that is, across the axis of symmetry; and this is the direction observed in practice (2). On the other hand, cavities formed near the bonded surfaces are in an isotropic stress field with no preferred direction. In practice, they propagate along the axis and thus do not lead directly to failure of the bonded structure.

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References

1. A.N.Gent and B.Park, Rubber Chem.Technol. 59, 77 (1986).
2. A.N.Gent and B.Park, J. Materials Sci. 19, 1947 (1984).
3. P.B.Lindley, "Engineering Design With Natural Rubber", 3rd. Ed., NR Technical Bulletin, Natural Rubber Producer's Research Association, London, 1970.
4. A.N.Gent and P.B.Lindley, Proc.Roy.Soc.(London) A249, 195 (1958).
5. A.E.Oberth and R.S.Bruenner, Trans.Soc.Rheol. 9, 165 (1965).
6. A.N.Gent and D.A.Tompkins, J.Appl.Phys. 40, 2520 (1969).
7. K-J.Bathe, "ADINA: A Finite Element Program for Automatic Dynamic
Nonlinear Incremental Analysis," Report No. 82448-1, Mass. Inst. Technol.,
Cambridge, Mass., 1977.

Table 1: Stiffness, $F/\delta DE$ for an incompressible elastic layer bonded between two spheres of diameter D and initial separation h .

h/D	$F/\delta DE$ from FEM	$F/\delta DE$ from approximate theory (1)	<u>Difference</u> (per cent)
0.01	40.77	40.90	0.32
0.02	20.916	21.01	0.45
0.05	8.704	8.904	2.24
0.1	4.519	4.749	4.84
0.2	2.399	2.579	6.96
0.5	1.122	1.171	4.13
1	0.648	0.643	-0.86
2	0.358	0.347	-3.32

Figure Legends

- Figure 1. Sketch of the model employed for FEM calculations.
- Figure 2. Calculated stiffness F/δ plotted against the value of Poisson's ratio ν , for various thicknesses h of the elastic layer relative to the sphere diameter D .
- Figure 3. Stiffness F/δ plotted against the layer thickness h , relative to the sphere diameter D . The open circles represent the results of FEM calculations for an incompressible material. The filled-in circles are experimentally-measured values for a silicone rubber layer, $E = 2$ MPa, (1). The full line represents the results of a previous approximate theory (1).
- Figure 4. Pressure P_c , developed at the center of an elastic layer, plotted against the value of Poisson's ratio ν , for various values of the layer thickness h , relative to the sphere diameter D . P_o denotes the mean applied stress, given by $4F/\pi D^2$.
- Figure 5. Pressure P_c , developed at the center of an incompressible elastic layer, plotted against the layer thickness h relative to the sphere diameter D . P_o denotes the mean applied stress, given by $4F/\pi D^2$. The full curve represents the predictions of a previous approximate theory (1). The points represent values calculated by FEM.
- Figure 6. Pressure P developed along the central axis, plotted

against the distance z from the center of an incompressible elastic layer of thickness h . The point $z = h/2$ is at the bonded interface. P_c denotes the pressure set up at the center of the layer.

Figure 7. Ratio of the radial stress σ_r to the axial stress σ_z , plotted against the distance z from the center of an incompressible elastic layer of thickness h .

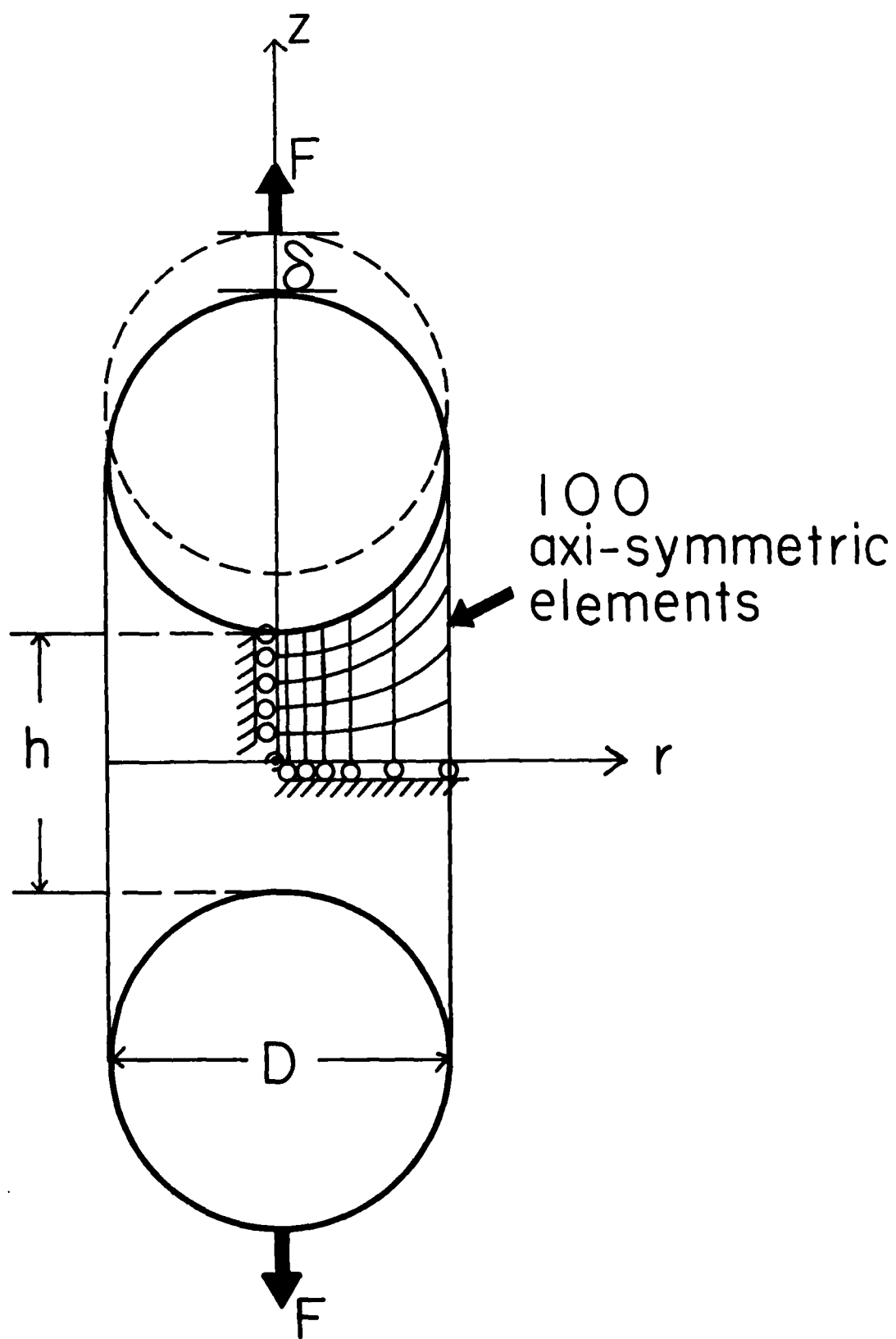


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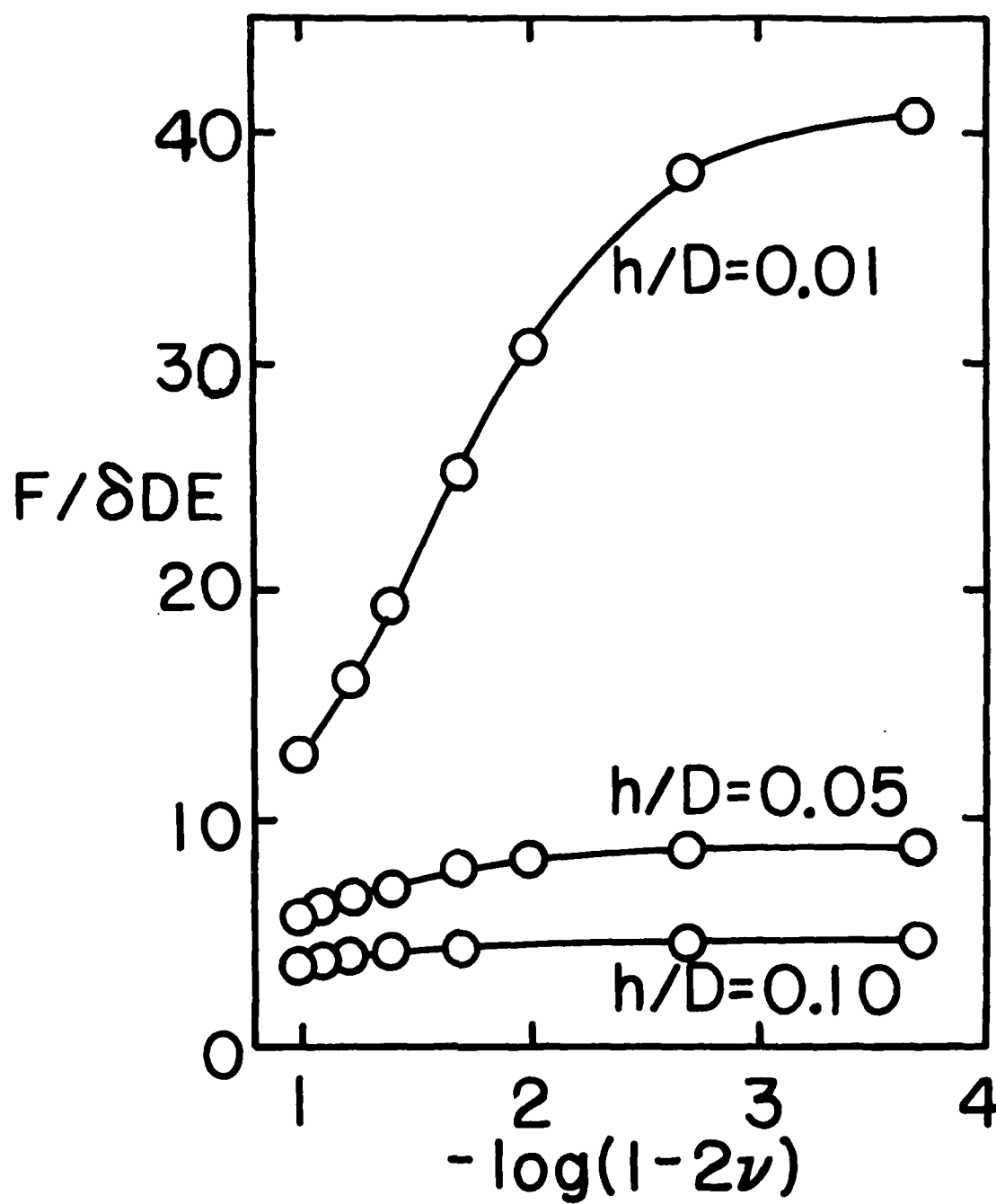


Figure 2

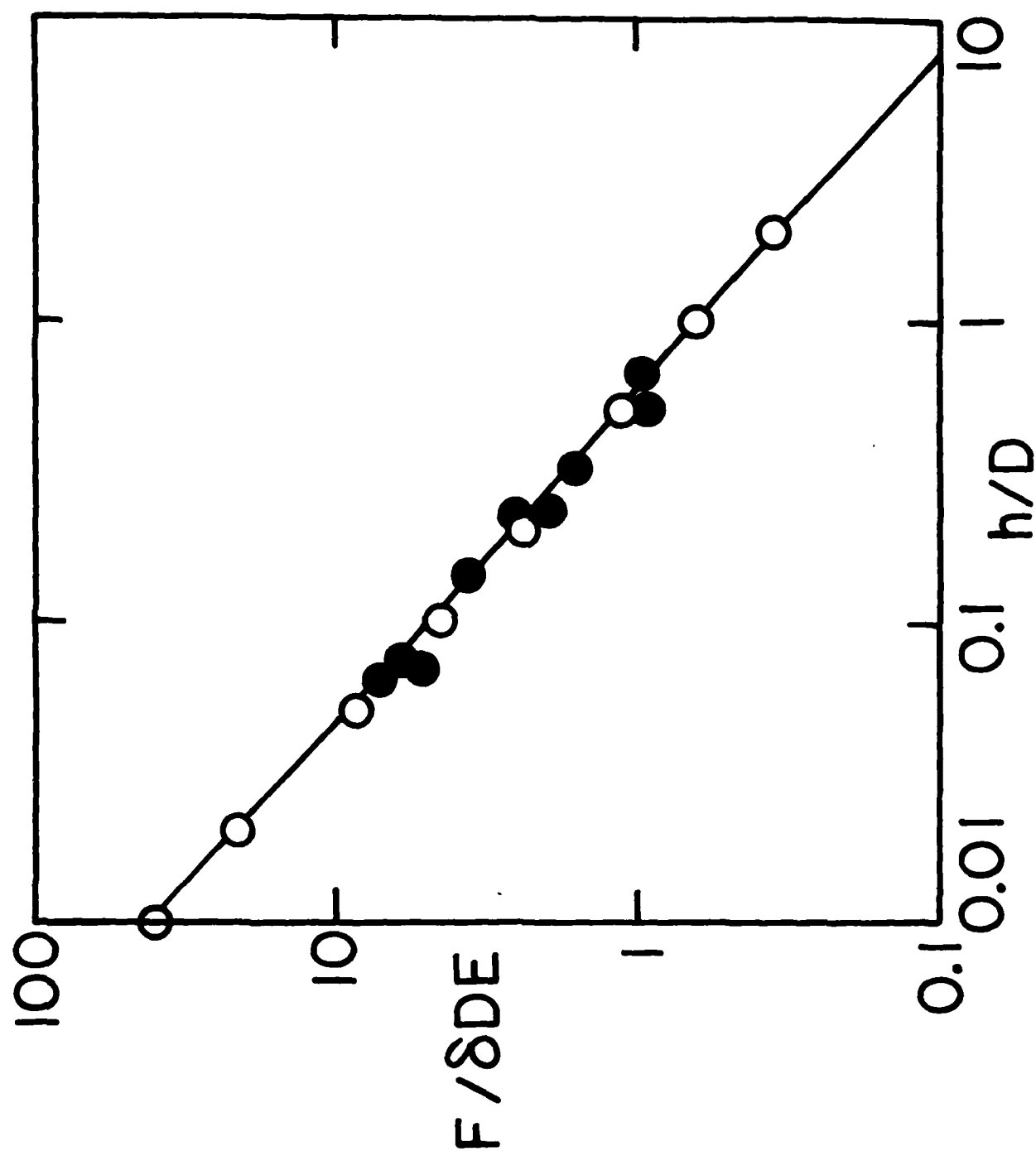


Figure 3

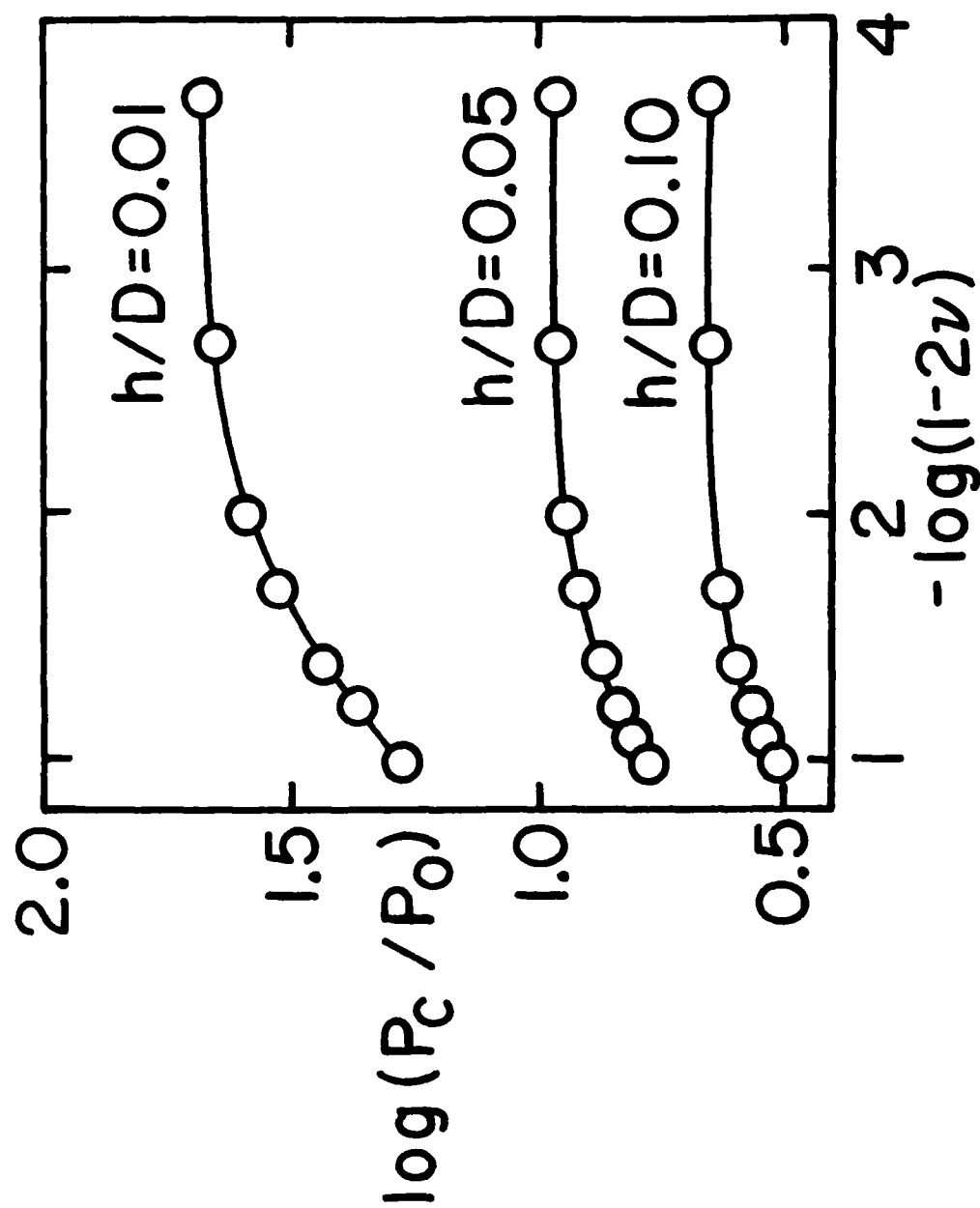


Figure 4

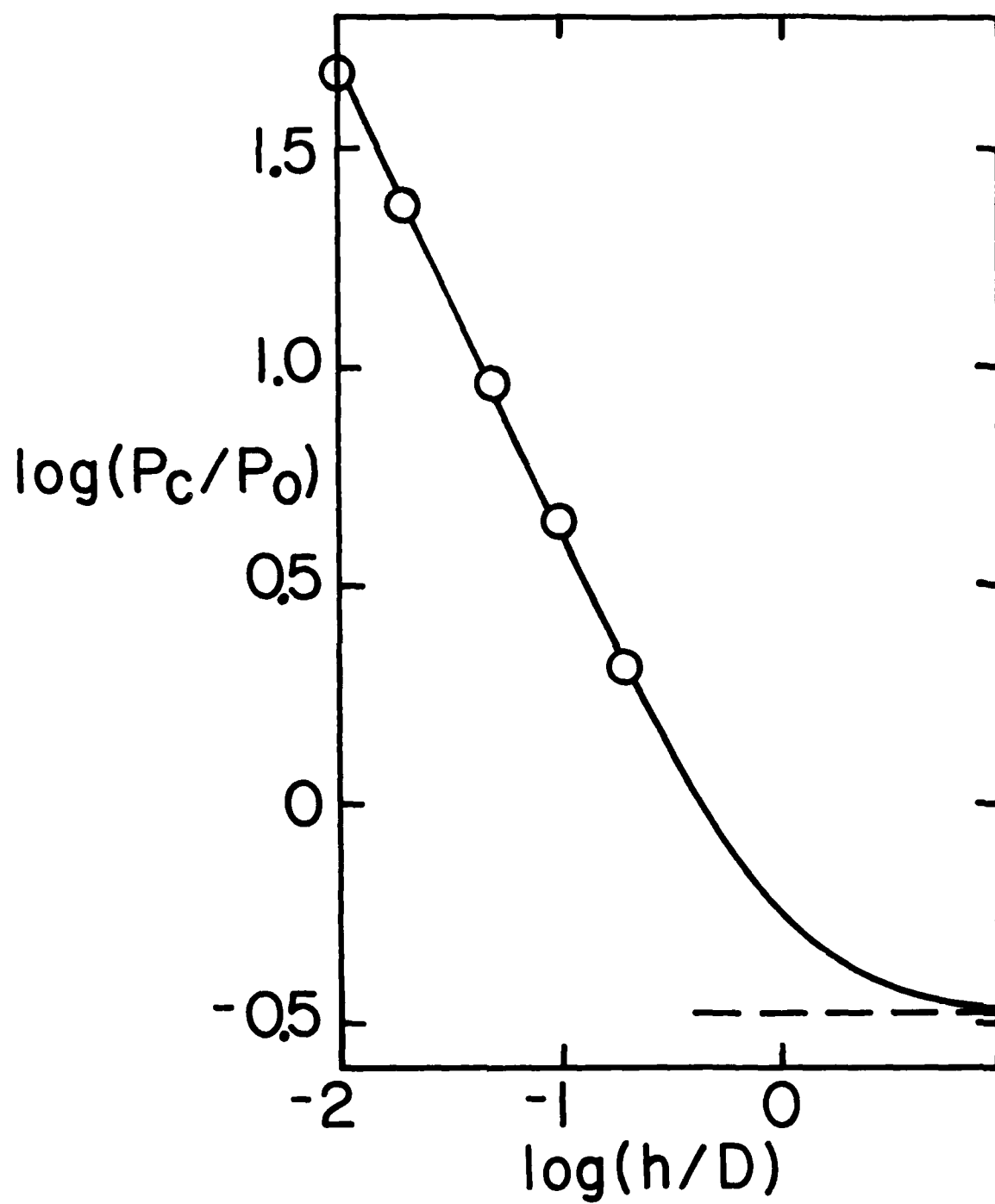


Figure 5

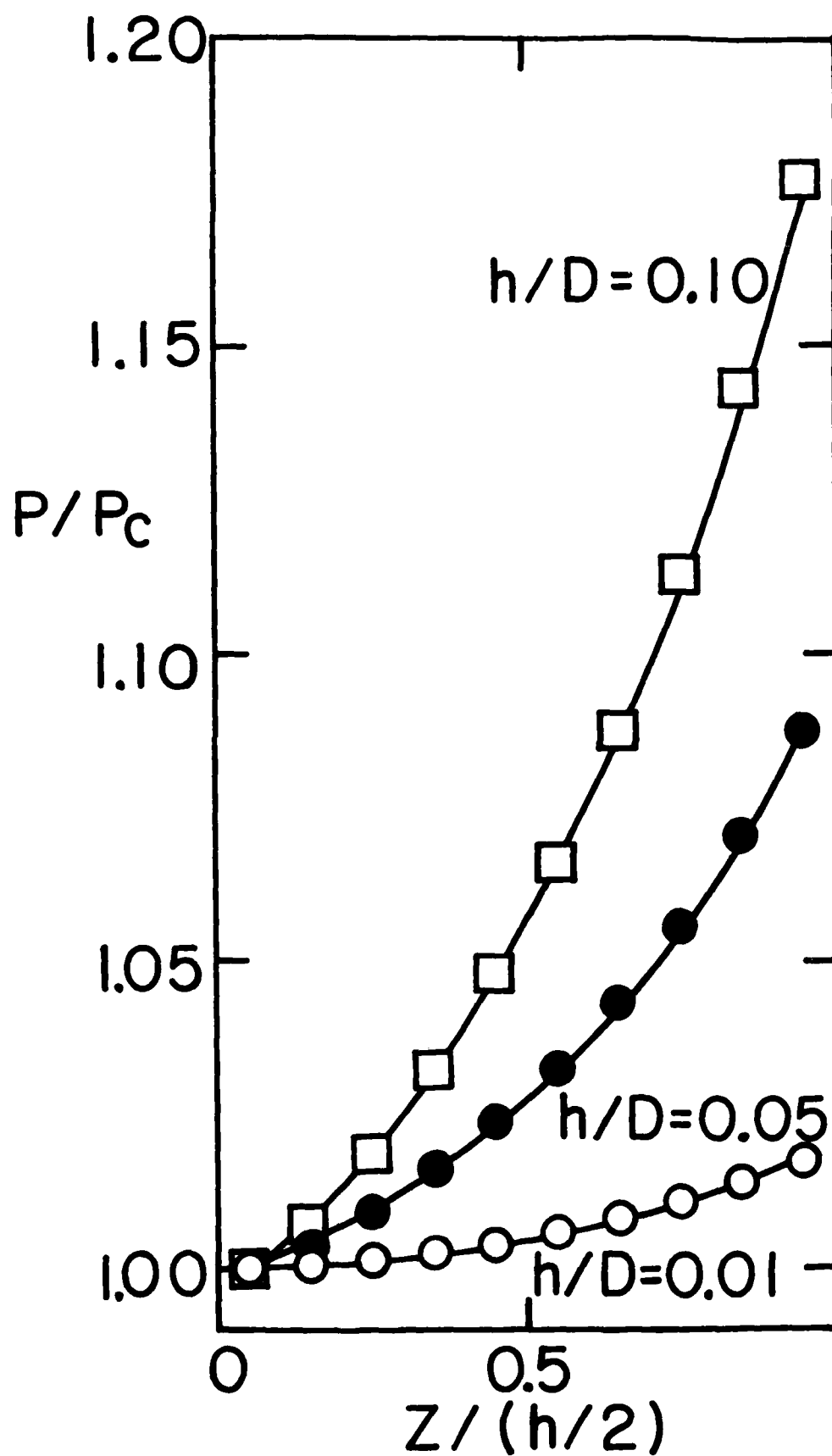


Figure 6

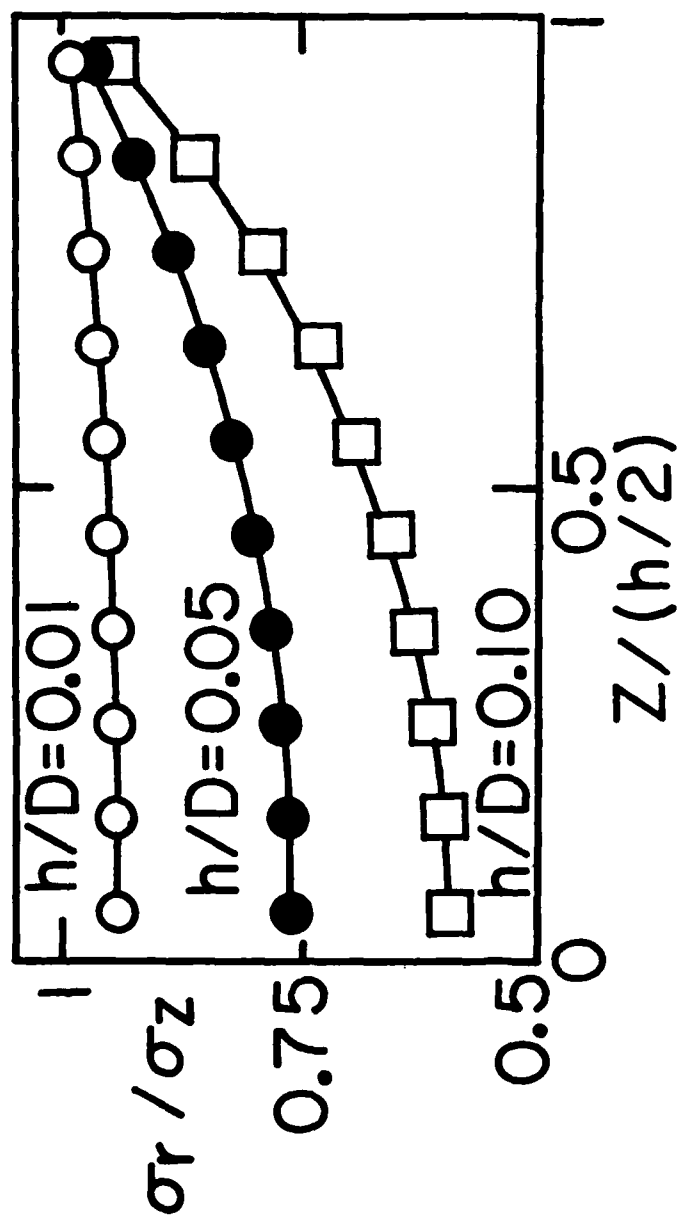


Figure 7

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