

UNCLASSIFIED
SECURITY CLASSIFICATION OF THIS PAGE (When Data Entered) REPORT DOCUMENT TION PAGE BEFORE COMPLETING FORM REPORT NUMBER 3. PECIPIENT'S CATALOG NUMBER 2. GOVT ACCESSION NO CI 77-59 TITLE (and Subtitle) 5. TYPE OF REPORT & PERIOD COVERED Progressive Failure of Advanced Composite Laminates Using the Finite Element Method Thesis PERFORMING ORG. REPORT NUMBER AUTHOR(s) 8. CONTRACT OR GRANT NUMBER(s) AD A 043 GARY E. BROWN 10. PROGRAM ELEMENT, PROJECT, TASK PERFORMING ORGANIZATION NAME AND ADDRESS AFIT Student at University of Utah, Salt Lake City, Utah 11. CONTROLLING OFFICE NAME AND ADDRESS 12. REPORT DATE March 1976 AFIT/CI Wright-Patterson AFB OH 45433 NUMBER OF PAGES 88 pages 15. SECURITY CLASS. (of this report) 14. MONITORING AGENCY NAME & ADDRESS(if different from Controlling Office) Unclassified DECLASSIFICATION/DOWNGRADING SCHEDULE 16. DISTRIBUTION STATEMENT (of this Report) Approved for Public Release; Distribution Unlimited I-77-59 17. DISTRIBUTION STATEMENT (of the abstract entered in Block 20, if different from 18. SUPPLEMENTARY GUESS, Captain, USAF Director of Information, AFIT 19. KEY WORDS (Continue on reverse side if necessary and identify by block number) 20. ABSTRACT (Continue on reverse side if necessary and identify by block number) Attached

DD 1 JAN 73 1473

EDITION OF 1 NOV 65 IS OBSOLETE

UNCLASSIFIED

012 200

#### INSTRUCTIONS FOR PREPARATION OF REPORT DOCUMENTATION PAGE

RESPONSIBILITY. The controlling DoD office will be responsible for completion of the Report Documentation Page, DD Form 1473, in all technical reports prepared by or for DoD organizations.

<u>CLASSIFICATION</u>. Since this Report Documentation Page, DD Form 1473, is used in preparing announcements, bibliographies, and data banks, it should be unclassified if possible. If a classification is required, identify the classified items on the page by the appropriate symbol.

#### COMPLETION GUIDE

General. Make Blocks 1, 4, 5, 6, 7, 11, 13, 15, and 16 agree with the corresponding information on the report cover. Leave Blocks 2 and 3 blank.

- Block 1. Report Number. Enter the unique alphanumeric report number shown on the cover.
- Block 2. Government Accession No. Leave Blank. This space is for use by the Defense Documentation Center.
- Block 3. Recipient's Catalog Number. Leave blank. This space is for the use of the report recipient to assist in future retrieval of the document.
- Block 4. Title and Subtitle. Enter the title in all capital letters exactly as it appears on the publication. Titles should be unclassified whenever possible. Write out the English equivalent for Greek letters and mathematical symbols in the title (see "Abstracting Scientific and Technical Reports of Defense-sponsored RDT/E,"AD-667 000). If the report has a subtitle, this subtitle should follow the main title, be separated by a comma or semicolon if appropriate, and be initially capitalized. If a publication has a title in a foreign language, translate the title into English and follow the English translation with the title in the original language. Make every effort to simplify the title before publication.
- Block 5. Type of Report and Period Covered. Indicate here whether report is interim, final, etc., and, if applicable, inclusive dates of period covered, such as the life of a contract covered in a final contractor report.
- <u>Block 6.</u> Performing Organization Report Number. Only numbers other than the official report number shown in Block 1, such as series numbers for in-house reports or a contractor/grantee number assigned by him, will be placed in this space. If no such numbers are used, leave this space blank.
- Block 7. Author(s). Include corresponding information from the report cover. Give the name(s) of the author(s) in conventional order (for example, John R. Doe or, if author prefers, J. Robert Doe). In addition, list the affiliation of an author if it differs from that of the performing organization.
- Block 8. Contract or Grant Number(s). For a contractor or grantee report, enter the complete contract or grant number(s) under which the work reported was accomplished. Leave blank in in-house reports.
- Block 9. Performing Organization Name and Address. For in-house reports enter the name and address, including office symbol, of the performing activity. For contractor or grantee reports enter the name and address of the contractor or grantee who prepared the report and identify the appropriate corporate division, school, laboratory, etc., of the author. List city, state, and ZIP Code.
- Block 10. Program Element, Project, Task Area, and Work Unit Numbers. Enter here the number code from the applicable Department of Defense form, such as the DD Form 1498, "Research and Technology Work Unit Summary" or the DD Form 1634. "Research and Development Planning Summary," which identifies the program element, project, task area, and work unit or equivalent under which the work was authorized.
- Block 11. Controlling Office Name and Address. Enter the full, official name and address, including office symbol, of the controlling office. (Equates to funding/sponsoring agency. For definition see DoD Directive 5200.20, "Distribution Statements on Technical Documents.")
  - Block 12. Report Date. Enter here the day, month, and year or month and year as shown on the cover.
  - Block 13. Number of Pages. Enter the total number of pages.
- Block 14. Monitoring Agency Name and Address (if different from Controlling Office). For use when the controlling or funding office does not directly administer a project, contract, or grant, but delegates the administrative responsibility to another organization.
- Blocks 15 & 15a. Security Classification of the Report: Declassification/Downgrading Schedule of the Report. Enter in 15 the highest classification of the report. If appropriate, enter in 15a the declassification/downgrading schedule of the report, using the abbreviations for declassification/downgrading schedules listed in paragraph 4-207 of DoD 5200.1-R.
- Block 16. Distribution Statement of the Report. Insert here the applicable distribution statement of the report from DoD Directive 5200.20, "Distribution Statements on Technical Documents."
- Block 17. Distribution Statement (of the abstract entered in Block 20, if different from the distribution statement of the report). Insert here the applicable distribution statement of the abstract from DoD Directive 5200.20, "Distribution Statements on Technical Documents."
- Block 18. Supplementary Notes. Enter information not included elsewhere but useful, such as: Prepared in cooperation with . . . Translation of (or by) . . . Presented at conference of . . . To be published in . . .
- Block 19. Key Words. Select terms or short phrases that identify the principal subjects covered in the report, and are sufficiently specific and precise to be used as index entries for cataloging, conforming to standard terminology. The DoD "Thesaurus of Engineering and Scientific Terms" (TEST), AD-672 000, can be helpful.
- Block 20. Abstract. The abstract should be a brief (not to exceed 200 words) factual summary of the most significant information contained in the report. If possible, the abstract of a classified report should be unclassified and the abstract to an unclassified report should consist of publicly- releasable information. If the report contains a significant bibliography or literature survey, mention it here. For information on preparing abstracts see "Abstracting Scientific and Technical Reports of Defense-Sponsored RDT&E," AD-667 000.

77-59 T

PROGRESSIVE FAILURE

OF ADVANCED COMPOSITE LAMINATES

USING THE FINITE ELEMENT METHOD

By
Gary Earl Brown

A thesis submitted to the faculty of the University of Utah in partial fulfillment of the requirements for the degree of

Master of Science

Department of Mechanical Engineering
University of Utah
March 1976



### UNIVERSITY OF UTAH GRADUATE SCHOOL

## FINAL READING APPROVAL

I have read the thesis of	GARY EARL	BROWN	in its
			ns, and bibliographic style are
consistent and acceptable;	(2) its illustra	ative materials	including figures, tables, and
charts are in place; and (	3) the final m	nanuscript is	satisfactory to the Supervisory
Committee and is ready for	or submission	to the Gradu	ate School.

To the Graduate Council of the University of Utah:

Approved for the Major Department

Robert F. Boehm Chairman/Dean

Approved for the Graduate Council

Sterling M. McMurrin Dean of the Graduate School

## UNIVERSITY OF UTAH GRADUATE SCHOOL

## SUPERVISORY COMMITTEE APPROVAL

of a thesis submitted by

GARY EARL BROWN

degree.	// //
3/3/76	Ralphy Luismer Ralphy Juismer
Date /	
	Chairman, Substvisory Committee

I have read this thesis and have found it to be of satisfactory quality for a master's degree.

3/3/76

William E. Mason Member, Supervisory Committee

### **ACKNOWLEDGEMENTS**

I would like to express my gratitude to Dr. Ralph J. Nuismer for suggesting this thesis topic and for his guidance and sharing his expert knowledge.

Appreciation is expressed to the Supervisory Committee members, Dr. Stephen R. Swanson and Dr. William E. Mason, for their time, helpful suggestions and comments.

In addition, the patience, understanding and encouragement of my wife, Mary, and my children, merits a special note of appreciation.

# TABLE OF CONTENTS

Acknowledgement		vii ix
INTRODUCTION		, 1
COMPOSITE FAILURE		6
Anisotropic Elasticity		7
Strain Displacement Relationships Laminate Constitutive Equations		12 16
Ply Failure Criteria		19
Modified Maximum Strain Theory Post Failure Constitutive Equations		19 20
Laminate Failure Criteria		24
NUMERICAL PROCEDURE		27
Finite Element Method for Plane Stress Analysis Solution Method for Nonlinear Material Properties.		
Initial Stress Process		32 33
SOLUTION OF PROBLEMS		36
Uniaxial Tension Specimens		36 36
RESULTS		36
Uniaxial Tension Specimens		
(0°/90°/90°/0°) <sub>s</sub> Laminate		40 51

DISCUSSION					56
APPENDIX A - THORNEL 300/5208 GRAPHITE-EPOXY PROF	PER1	TIE	S.		60
APPENDIX B - COMPUTER PROGRAM DESCRIPTION					61
APPENDIX C - COMPUTER PROGRAM VARIABLES					65
APPENDIX D - COMPUTER PROGRAM INPUT DATA					67
APPENDIX E - COMPUTER PROGRAM					70
LIST OF REFERENCES					86
VITA					89



# LIST OF FIGURES

Figure		
1	Unidirectional lamina	7
2	Positive rotation of principal material axes from arbitrary axes	9
3	Geometry of deformation in the xz plane	13
4	Plane stress triangular element	28
5	Finite element grid, uniaxial tension test	37
6	Finite element grid, circular hole in uniaxial tension.	39
7	Stress vs strain, $(0^\circ)_S$ laminate in uniaxial tension	41
8	Stress vs strain, $(90^\circ)_{\rm S}$ laminate in uniaxial tension .	42
9	Stress vs strain, $(0^{\circ}/90^{\circ}/90^{\circ}/0^{\circ})_{s}$ laminate in uniaxial tension	43
10	Stress vs strain, $(90^{\circ}/+45^{\circ}/0^{\circ})_{s}$ laminate in uniaxial tension	44
11	Stress vs strain, $(90^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$ laminate in uniaxial tension	45
12	$(0^{\circ}/90^{\circ}/90^{\circ}/0^{\circ})_{S}$ circular hole partial failure at .009 in (.023 cm) displacement load	46
13	$(0^{\circ}/90^{\circ}/90^{\circ}/0^{\circ})_{S}$ circular hole partial failure at .012 in. (.030 cm) displacement load	47
14	$(0^\circ/90^\circ/90^\circ/0^\circ)_S$ circular hole partial failure at .015 in. (.038 cm) displacement load	48
15	$(0^\circ/90^\circ/90^\circ/0^\circ)_S$ circular hole partial failure at .018 in. (.046 cm) displacement load	49
16	(0°/90°/90°/0°) <sub>s</sub> circular hole, complete failure at .021 in. (.053 cm) displacement load	50

17	$(0^{\circ}/\pm 45^{\circ}/90^{\circ})$ circular hole partial .008 in. (.020 cm) displacement load.	failure	at 		52
18	$(0^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$ circular hole partial .012 in. (.030 cm) displacement load.	failure	at 		53
19	$(0^{\circ}/\pm~45^{\circ}/90^{\circ})_{\rm S}$ circular hole partial .016 in. (.041 cm) displacement load.	failure	at 		54
20	$(0^{\circ}/\pm 45^{\circ}/90^{\circ})_{S}$ circular hole, complet .020 in. (.051 cm) displacement load.	te failu	re at		55

## NOMENCLATURE

[A] = system stiffness matrix = coefficient in area coordinate equation ai  $[B_{ij}]$ = laminate coupling stiffness matrix [B] = strain-displacement matrix = coefficient in area coordinate equation bi = coefficient in area coordinate equation Ci [D<sub>ii</sub>] = laminate inplane stiffness matrix E11 = Young's modulus in the 1 direction E22 = Young's modulus in the 2 direction = superscript indicates element value = nodal force array {F} = lamina shear modulus G12 i,j,k,m,n= dummy indices [K] = element stiffness matrix  $M_x$ ,  $M_y$ ,  $M_{xy}$ = moment resultants  $N_x$ ,  $N_y$ ,  $N_{xy}$ = stress resultants {PF} = pseudo nodal force vector PFmax = maximum pseudo nodal force = laminate stiffness matrix in material coordinate [Q] system  $[\overline{Q}]$ = laminate stiffness matrix in arbitrary coordinate  $\mathsf{S}_{\varepsilon}$ = maximum allowable shear strain

[T] = transformation matrix = laminate thickness t = invariant material constants for lamina U<sub>1-5</sub> Ui = nodal force, x-direction U = strain energy = displacement, x-direction ٧i = nodal force, y-direction = displacement, y-direction = work = displacement, z-direction W  $\boldsymbol{x}_{\epsilon t}$ = maximum allowable tensile strain, 1 direction  $X_{\epsilon C}$ = maximum allowable compressive strain, 1 direction x,y,z= arbitrary coordinate system  $Y_{\epsilon t}$ = maximum allowable tensile strain, 2 direction Υ<sub>εc</sub> = maximum allowable compressive strain, 2 direction = constant  $\alpha_{i}$ Δ = indicates change in variable = shear strain Υ {8} = displacement vector = normal strain ε θ = angular displacement = curvature K = normal stress σ = shear stress = major Poisson's ratio <sup>ν</sup>12

<sup>ν</sup>21

= minor Poisson's ratio

 $\triangle$ 

= area, triangular element

#### **ABSTRACT**

In the study of fiber-reinforced resin composites, the analysis of the progressive failure of a laminate with a stress concentration subjected to plane stress poses a very interesting but complex problem. This thesis approaches this problem by using the finite element method to examine the progressive failure of symmetrical laminates.

A modified maximum strain failure theory is proposed and a finite element computer program developed that accounts for progressive failure. A computer analysis of several unnotched laminate tensile specimens, with lamina at various angles, was made and these results are compared with experimental data.

Circular hole tensile specimens with  $(0^\circ/90^\circ/90^\circ/0^\circ)_s$  and  $(0^\circ/\pm 45^\circ/90^\circ)_s$  lamina were also investigated, and the progressive failure through the finite element grid presented. The ultimate failure loads of the circular hole specimens are compared with experimental data. Material properties used were those for Thornel 300/5208 Graphite-Epoxy.

Although the results obtained cannot be considered conclusive for all cases, they do compare favorably with experimental data for the unnotched specimens. The ultimate failure loads of the hole specimens were somewhat higher than those obtained experimentally.

### INTRODUCTION

The word "composite" in composite material signifies that two or more materials are combined on a macroscopic scale to form a useful material. The advantage of composites is that the materials can be combined in ways that usually exhibit the best qualities of their constituents and often some qualities that neither constituent possesses. Some properties that may be improved by use of a composite material are strength, stiffness, weight, corrosion resistance, fatigue life, and thermal properties. <sup>1</sup>

Composite materials have been used for centuries. When the first composite was used is unknown, but recorded history contains references to various forms of composite materials. For example, the Egyptians used laminated wood as early as 2780 B.C., and the Israelites added chopped straw to the manufacture of bricks in 800 B.C.<sup>2</sup> A short time thereafter, the Mongol bow was developed from a composite of animal tendons, wood and silk bonded together with an adhesive. Still later, laminated structures appeared in the Damascus gun barrels and Japanese ceremonial swords.<sup>3</sup>

More recently, fiber-reinforced resin composites that have a high strength-to-weight and stiffness-to-weight ratio have become important in weight sensitive applications such as aircraft and space vehicles. Some examples of these modern applications of fiber-resin composites are: an AT-6C aircraft with reinforced plastic fuselage

built in 1943, helicopter rotor blades specifically designed to reduce vibration and withstand torsional loads, hocket motor cases, space vehicle structural components, and presently an increasing number of uses such as fuselage and stabilizer components on the F-111, F-14, F-15, and F-16 aircraft. Through the use of structural compounds made of composite materials, strength-to-weight ratios have been increased 100 percent over that of comparable metal structures. The impact of composites on jet engine performance may be even more dramatic, where an 800 percent increase in the thrust-to-weight index appears possible.

The superior strength-to-weight ratio of these fibrous composites is related to the failure mechanisms of homogeneous materials where, generally, the actual strength is considerably lower than the theoretical atomic strength. The reason for this strength difference is the formulation and movement of dislocations in the homogeneous material. By forming a material into thin whiskers or fibers with a small cross section, conformity in the microstructure is enhanced, the probability of internal flaws is reduced, and the formation and movement of dislocations restricted, making it possible for the fiber or whisker to approach its theoretical strength. Composite sheets or "lamina" with high longitudinal strength are formed by imbedding many of these high strength fibers longitudinally in a suitable matrix material. A composite "laminate" with the desired strength and stiffness properties may then be formed by combining layers of lamina together at various orientations.

Investigation of lamina strength has generally been approached from both micromechanical and macromechanical levels. 1,9,10 The

micromechanical approach, which treats a composite material as a heterogeneous continuum, has been used for simple lamina models. Although theoretically justifiable, this approach has a major limitation in that the analysis required is extremely complex, and therefore, limited to very simple geometries. 31

In general, macromechanical prediction of lamina failure has been approached from one of the following three theories: the maximum strain theory, maximum stress theory, or maximum work theory. 12,14 Of these three theories, the maximum work approach has been proven to be the most accurate when compared with experimental data. 1,12,13 However, this theory does not easily lend itself to the analysis of progressive failure because the damage to the composite cannot be described and put into post-failure relations. The maximum strain and maximum stress theories are well suited for a progressive failure type analysis, but the accuracy of these theories deteriorates when the fibers are at an angle between 15 and 60 degrees to an applied uni-axial load. The reason for this loss of accuracy at intermediate angles is probably due to not considering the interactive effect on failure of combined shear and tension.

Failure of unnotched laminates has been approached by combining plies through lamination theory and applying a lamina failure theory to each ply. In doing so, the disadvantages of ply failure theories are carried through to the laminate. In addition, failure of one lamina as a failure criterion for the laminate is usually too conservative. Maximum work or distortional energy applications to laminate failure, such as that by Tasi-Wu, have given good results for individual laminates, but new properties must be obtained for each

new laminate. Sendeckyj,<sup>29</sup> using the method of Sandhu,<sup>36</sup> successfully used lamina stress-strain data to predict the nonlinear response of angle and multi-ply laminates in uniaxial tension. The failure predictions were less successful, however. The success of this approach to progressive failure of notched laminates is still to be determined.

Because of the complexity of the micromechanical approach, macromechanical principles are usually employed to determine laminate behavior in the presence of notches. The anisotropy of the laminate makes the failure properties due to stress concentrations of particular interest in that the stress concentration factor can be considerably higher for composites than for isotropic materials. In addition, a hole and crack size effect has been observed on laminate strength. Macromechanical studies of notched laminate failure have used models such as the 'inherent flaw model' for holes by Waddoups<sup>35</sup> and an 'average' or 'point' stress approach for holes and cracks by Nuismer and Whitney.<sup>30</sup> However, both studies neglect the load-path dependent damage or progressive failure of the laminate. In doing so, none can be expected to have general applicability, especially when biaxial loading is considered.

The purpose of this thesis is to study progressive failure of notched laminates subjected to in plane loads.

This thesis:

- presents a modified maximum strain theory for individual plies and develops the post failure constitutive relations for the ply;
- (2) develops an incremental finite element program which uses

- laminated plate theory to account for stiffness changes in the laminate due to failure in the plies;
- (3) investigates the stress-strain behavior to failure of several unnotched laminates (under uniaxial tension loads) and compares the results with experimental data;
- (4) traces the progressive failure of two laminates containing circular holes and compares predicted failure loads with experimental data.

## COMPOSITE FAILURE

# Anisotropic Elasticity

With the advent and increased usage of fibers such as graphite in composite materials, the assumption that the material is isotropic is no longer valid. Graphite fibers are highly anisotropic, with the longitudinal stiffness being an order of magnitude greater than the transverse stiffness. Then, in order to analyze such fiber-reinforced composites, anisotropic elasticity must be employed.

For a three dimensional stress state, the generalized Hooke's Law for an anisotropic material is given by:

$$\begin{pmatrix}
\sigma_{1} \\
\sigma_{2} \\
\sigma_{3} \\
\tau_{23} \\
\tau_{12} \\
\tau_{12}
\end{pmatrix} = \begin{bmatrix} Q_{ij} \end{bmatrix} \begin{pmatrix}
\varepsilon_{1} \\
\varepsilon_{2} \\
\varepsilon_{3} \\
\gamma_{23} \\
\gamma_{31} \\
\gamma_{12}
\end{pmatrix} (1)$$

where the stiffness matrix  $Q_{ij}$  is symmetrical with 21 independent constants.  $^{1}$ 

If there are two orthogonal planes of material property symmetry, such as parallel and perpendicular to the fibers in a unidirectional fiber composite, symmetry will also exist relative to a third mutually orthogonal plane, and the material is said to be orthotropic. The stress-strain relations for an orthotropic material in a coordinate

system aligned with the principal material directions, or parallel and perpendicular to the fiber direction, becomes:

$$\begin{pmatrix} \sigma_{1} \\ \sigma_{2} \\ \sigma_{3} \\ \tau_{23} \\ \tau_{31} \\ \tau_{12} \end{pmatrix} = \begin{pmatrix} \overline{Q}_{11} & \overline{Q}_{12} & \overline{Q}_{13} & 0 & 0 & 0 \\ \overline{Q}_{12} & \overline{Q}_{22} & \overline{Q}_{23} & 0 & 0 & 0 \\ \overline{Q}_{13} & \overline{Q}_{23} & \overline{Q}_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & \overline{Q}_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & \overline{Q}_{55} & 0 \\ 0 & 0 & 0 & 0 & 0 & \overline{Q}_{66} \end{pmatrix} \begin{pmatrix} \varepsilon_{1} \\ \varepsilon_{2} \\ \varepsilon_{3} \\ \gamma_{23} \\ \gamma_{31} \\ \gamma_{12} \end{pmatrix}$$
 (2)

where the stiffness matrix has been reduced to nine independent constants.

## Lamina Constitutive Relationships

One pertinent assumption in establishing the constitutive or stress-strain relationships for the lamina of a laminated composite is that a lamina, when in a composite, is in a state of plane stress. For a state of plane stress, and with the lamina in the 1-2 plane as shown in Figure 1, the following stresses are assumed zero:

$$\sigma_3 = 0, \quad \tau_{23} = 0, \quad \tau_{13} = 0$$
 (3)

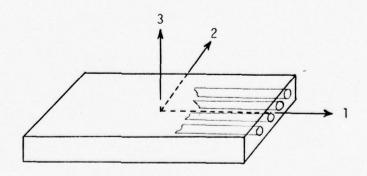


Figure 1. Unidirectional Lamina

By sutstituting into Equation (2), the stress-strain relation becomes:

$$\begin{pmatrix} \sigma_1 \\ \sigma_2 \\ \tau_{12} \end{pmatrix} = \begin{bmatrix} \overline{Q}_{11} & \overline{Q}_{12} & 0 \\ \overline{Q}_{12} & \overline{Q}_{22} & 0 \\ 0 & 0 & \overline{Q}_{66} \end{bmatrix} \quad \begin{pmatrix} \varepsilon_1 \\ \varepsilon_2 \\ \gamma_{12} \end{pmatrix} \tag{4}$$

where the components of the stiffness matrix for the orthotropic lamina given in terms of engineering constants are:

$$Q_{11} = E_{11}/(1-v_{12} v_{21})$$

$$Q_{22} = E_{22}/(1-v_{12} v_{21})$$

$$Q_{12} = v_{21}E_{11}/(1-v_{12} v_{21}) = v_{12}E_{22}/(1-v_{12} v_{21})$$

$$Q_{66} = G_{12}$$
(5)

There are now four independent constants:  $E_{11}$ ,  $E_{22}$ ,  $G_{12}$  and  $v_{12}$ , which are the elastic moduli in the 1 and 2 directions, the shear modulus and the major Poisson's ratio, respectively. The major and minor Poisson's ratios are related by:

$$v_{21}E_{11} = v_{12}E_{22}$$
 (6)

where the major Poisson's ratio,  $v_{12}$ , is the ratio of strain in the 2 direction to strain in the 1 direction for a load in the one direction. To tailor a material with the proper stiffness and strength in various directions, unidirectional laminae are usually put together with fibers running in several different directions so that the lamina principal axes are not coincident with the reference axes of the laminate. When this occurs, the constitutive relations for each lamina must be transformed to the laminate reference axes in order to

determine the laminate constitutive relationship. The transformation relationship for stress between an arbitrary x-y axes and the primary 1-2 axes, as shown in Figure 2, is

$$\begin{pmatrix} \sigma_1 \\ \sigma_2 \\ \tau_{12} \end{pmatrix} = [T_{ij}] \begin{pmatrix} \sigma_x \\ \sigma_y \\ \tau_{xy} \end{pmatrix}$$
(7)

while that for strain is

where the transformation matrix  $\boldsymbol{T}_{i\,j}$  is:

$$[T_{ij}] = \begin{bmatrix} m^2 & n^2 & 2mn \\ n^2 & m^2 & -2mn \\ -mn & mn & m^2 - n^2 \end{bmatrix}$$
 (9)

 $m = \cos \theta$  $n = \sin \theta$ 

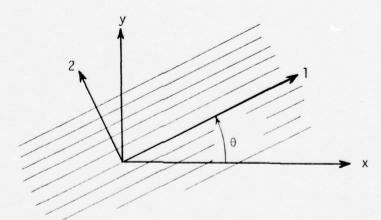


Figure 2. Positive rotation of principal material axes from arbitrary xy axes

In the same way, the primary stress and strain relations are referenced to the xy axes by:

$$\begin{pmatrix} \sigma_{x} \\ \sigma_{y} \\ \tau_{xy} \end{pmatrix} = [T_{ij}]^{-1} \begin{pmatrix} \sigma_{1} \\ \sigma_{2} \\ \tau_{12} \end{pmatrix}$$
(10)

$$\begin{cases}
\varepsilon_{x} \\
\varepsilon_{y} \\
\frac{\gamma_{xy}}{2}
\end{cases} = \left[T_{i,j}\right]^{-1} \begin{cases}
\varepsilon_{1} \\
\varepsilon_{2} \\
\frac{\gamma_{12}}{2}
\end{cases}$$
(11)

where T  $_{ij}$  inverse is obtained by substituting a negative angle  $\theta$  for the positive angle  $\theta$  in the T matrix. Thus, T inverse becomes:

$$[T_{ij}]^{-1} = \begin{bmatrix} m^2 & n^2 & -2mn \\ n^2 & m^2 & 2mn \\ mn & -mn & m^2 - n^2 \end{bmatrix}$$

$$m = \cos \theta$$

$$n = \cos \theta$$

Knowing the orthotropic lamina material properties and referencing them to the xy axes,  $\theta$  is measured in the negative direction. Then T becomes  $T^{-1}$  and the xy stress-strain relationship obtained from Equations (4), (8), and (10) is

where  $Q_{\mbox{ij}}$  is the orthotropic lamina stiffness from Equation (4).

Denoting the lamina stiffness with respect to the xy axes as  $\overline{\mathbb{Q}}_{\mathbf{i}\mathbf{j}},$  then

$$\begin{pmatrix} \sigma_{\mathbf{x}} \\ \sigma_{\mathbf{y}} \\ \tau_{\mathbf{x}\mathbf{y}} \end{pmatrix} = \left[ \overline{Q}_{\mathbf{i}\mathbf{j}} \right] \begin{pmatrix} \varepsilon_{\mathbf{x}} \\ \varepsilon_{\mathbf{y}} \\ \gamma_{\mathbf{x}\mathbf{y}} \end{pmatrix}$$
 (14)

where

$$[\overline{Q}_{ij}] = [T_{ij}] [Q_{ij}] [T_{ij}]^{-1}$$
(15)

Upon multiplication, the terms of the  $\overline{\mathbb{Q}}_{i,j}$  matrix become:

$$\overline{Q}_{11} = Q_{11}^{m^4} + 2(Q_{12} + 2Q_{66}) m^2 n^2 + Q_{22}^{m^4}$$

$$\overline{Q}_{12} = (Q_{11} + Q_{22} - 4Q_{66}) m^2 n^2 + Q_{12}^{m^4} + n^4)$$

$$\overline{Q}_{13} = (Q_{11} + Q_{12} + 2Q_{66}) m^3 n - (Q_{12} - Q_{22} + 2Q_{66}) mn^3$$

$$\overline{Q}_{22} = Q_{11}^{m^4} + 2(Q_{12} + 2Q_{66}) m^3 n^2 + Q_{22}^{m^4}$$

$$\overline{Q}_{23} = (-Q_{11} + Q_{12} + 2Q_{66}) mn^3 - (Q_{12} - Q_{22} + 2Q_{66}) m^3 n$$

$$\overline{Q}_{33} = (Q_{11} + Q_{22} - 2Q_{66}) m^2 n^2 + Q_{66}^{m^2} + n^2)$$
(16)

A more convenient form of the transformed lamina stiffness, in terms of invariants as given by Tsai and Pagano<sup>3</sup> and used later in the computer analysis, is:

$$\overline{Q}_{11} = U_1 + U_2 \cos(2\theta) + U_3 \cos(4\theta) 
\overline{Q}_{12} = U_4 - U_3 \cos(4\theta) 
\overline{Q}_{13} = \frac{1}{2} U_2 \sin(2\theta) + U_3 \sin(4\theta) 
\overline{Q}_{22} = U_1 - U_2 \cos(2\theta) + U_3 \cos(4\theta) 
\overline{Q}_{23} = \frac{1}{2} U_2 \sin(2\theta) - U_3 \sin(4\theta) 
\overline{Q}_{33} = U_5 - U_3 \cos(4\theta)$$
(17)

where

$$U_{1} = \frac{1}{8} (3Q_{11} + 3Q_{22} + 2Q_{12} + 4Q_{66})$$

$$U_{2} = \frac{1}{2} (Q_{11} - Q_{22})$$

$$U_{3} = \frac{1}{8} (Q_{11} + Q_{22} - 2Q_{12} - 4Q_{66})$$

$$U_{4} = \frac{1}{8} (Q_{11} + Q_{22} + 6Q_{12} - 4Q_{66})$$

$$U_{5} = \frac{1}{8} (Q_{11} + Q_{22} - 2Q_{12} + 4Q_{66})$$
(18)

## Laminated Plate Theory

## Strain Displacement Relationships

A laminate is composed of several orthotropic layers. As such, the description of the behavior of a single lamina, as previously discussed, forms the basis or building block with which the behavior of a laminate may be described. Equation (14) gives the constitutive relationship for a lamina with respect to an arbitrary xy coordinate system. Considering the arbitrary xy axes to be oriented with the laminate axes, Equation (14) can be thought of as a stress-strain relationship for the k<sup>th</sup> layer of a multi-layered laminate and may be written as

$$\{\sigma\}_{k} = \left[\overline{Q}\right]_{k} \{\epsilon\}_{k} \tag{19}$$

Knowing the variation of stress and strain through the laminate thickness is essential to the definition of the extensional and bending stiffness of a laminate. The laminate is assumed to consist of layers of perfectly bonded laminae, such that the displacements are continuous across lamina boundaries and one lamina cannot slip

relative to another. With this assumption, and if the laminate is thin, it may be assumed that a line originally straight and perpendicular to the middle surface of the laminate will remain straight and perpendicular to the middle when the laminate is extended and bent.

Considering a section of laminate in the xy plane deformed due to some loading, as shown in Figure 3, the geometrical midplane undergoes some displacement,  $\mathbf{u}_0$ , in the x-direction. With the above assumption, the line ABD remains straight and normal to the deformed

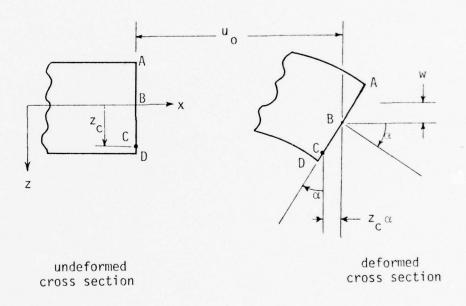


Figure 3. Geometry of deformation in the xz plane

midplane and the displacement in the x-direction of any point, C, on the normal ABD is given by the linear relationship $^3$ 

$$u_{c} = u_{0} - z_{c}^{\alpha} \tag{20}$$

where  ${\bf z}_{\bf C}$  is the z coordinate of the point C and  $\alpha$  is the slope of ABD with respect to the original vertical line. Also, under deformation, line ABD remains perpendicular to the middle surface so that the slope of the laminate surface in the x-direction is

$$\alpha = \frac{\partial \mathbf{w}}{\partial \mathbf{x}} \tag{21}$$

where w is the displacement in the z-direction. Substituting Equation (21) into Equation (20), the displacement, u, at any point, z, through the laminate thickness is

$$u = u_0 - z \frac{\partial w}{\partial x}$$
 (22)

By similar reasoning, the displacement, v, in the y-direction is

$$v = v_0 - z \frac{\partial w}{\partial y} \tag{23}$$

The assumptions thus far are equivalent to ignoring the shearing strains in planes perpendicular to the middle surface, that is,  $\gamma_{xz}$ ,  $\gamma_{yz}$  = 0. Also, the line ABD is assumed to have constant length so that  $\varepsilon_z$  = 0. These assumptions, known as the Kirchhoff-Love hypothesis, reduce the strain-displacement relationships to 1

$$\begin{array}{rcl}
\varepsilon_{X} & = & \frac{\partial u}{\partial x} \\
\varepsilon_{Y} & = & \frac{\partial v}{\partial y} \\
\Upsilon_{XY} & = & \frac{\partial u}{\partial y} + \frac{\partial v}{\partial x}
\end{array} \tag{24}$$

Differentiating Equations (22) and (23) and substituting into Equation (24), the strains become

$$\varepsilon_{x} = \frac{\partial u_{o}}{\partial x} - z \frac{\partial^{2} w}{\partial x^{2}}$$

$$\varepsilon_{y} = \frac{\partial v_{o}}{\partial y} - z \frac{\partial^{2} w}{\partial y^{2}}$$

$$\gamma_{xy} = \frac{\partial u_{o}}{\partial y} + \frac{\partial v_{o}}{\partial x} - 2z \frac{\partial^{2} w}{\partial x \partial y}$$
(25)

or

$$\begin{pmatrix}
\varepsilon_{X} \\
\varepsilon_{y} \\
\gamma_{xy}
\end{pmatrix} = \begin{pmatrix}
\varepsilon_{X}^{0} \\
\varepsilon_{y}^{0} \\
\gamma_{xy}^{0}
\end{pmatrix} + z \begin{pmatrix}
\kappa_{X} \\
\kappa_{y} \\
\kappa_{xy}
\end{pmatrix} \tag{26}$$

where the middle surface strains are

$$\begin{pmatrix}
\varepsilon_{x}^{0} \\
\varepsilon_{y}^{0} \\
\gamma_{xy}^{0}
\end{pmatrix} = \begin{pmatrix}
\frac{\partial u_{0}}{\partial x} \\
\frac{\partial v_{0}}{\partial y} \\
\frac{\partial v_{0}}{\partial y} + \frac{\partial v_{0}}{\partial x}
\end{pmatrix}$$
(27)

and the middle surface curvatures are1

$$\begin{pmatrix}
\kappa_{\mathbf{X}} \\
\kappa_{\mathbf{y}}
\end{pmatrix} = \begin{pmatrix}
\frac{\partial^{2} \mathbf{w}}{\partial \mathbf{x}^{2}} \\
\frac{\partial^{2} \mathbf{w}}{\partial \mathbf{y}^{2}} \\
2\frac{\partial^{2} \mathbf{w}}{\partial \mathbf{x} \partial \mathbf{y}}
\end{pmatrix} (28)$$

By substitution of the strain variation through the thickness, Equation (26) into Equation (19), the stresses in the  $k^{th}$  layer can be expressed in terms of the laminate middle surface strains and

curvatures as

$$\begin{pmatrix}
\sigma_{x} \\
\sigma_{y} \\
\varepsilon_{xy}
\end{pmatrix}_{k} = \begin{bmatrix}
\overline{Q}_{11} & \overline{Q}_{12} & \overline{Q}_{13} \\
\overline{Q}_{12} & \overline{Q}_{22} & \overline{Q}_{23} \\
\overline{Q}_{13} & \overline{Q}_{23} & \overline{Q}_{33}
\end{bmatrix}_{k} \begin{pmatrix}
\varepsilon_{x}^{0} \\
\varepsilon_{y}^{0} \\
\varepsilon_{xy}^{0}
\end{pmatrix} + z \begin{pmatrix}
\kappa_{x} \\
\kappa_{y} \\
\kappa_{xy}
\end{pmatrix} \qquad (29)$$

## Laminate Constitutive Equations

Since the  $\overline{\mathbb{Q}}_{ij}$  can be different for each layer of the laminate, the stress variation through the laminate is not necessarily linear even though the strain variation is linear. To investigate these non-linear stresses, the resultant laminate forces and moments, denoted by N and M respectively, are obtained by integration of the stresses in each layer of lamina through the laminate thickness. For example,

$$N_{x} = \int_{\frac{t}{2}}^{\frac{t}{2}} \sigma_{x} dz$$

$$M_{x} = \int_{\frac{t}{2}}^{\frac{t}{2}} \sigma_{x} zdz$$
(30)

where t is the total laminate thickness.

The total force and moment resultants for a n-layered laminate may then be defined as

and

$$\begin{pmatrix} M_{x} \\ M_{y} \\ M_{xy} \end{pmatrix} = \int_{-\frac{t}{2}}^{\frac{t}{2}} \begin{pmatrix} \sigma_{x} \\ \sigma_{y} \\ \tau_{xy} \end{pmatrix} z dz$$
(32)

or, using Equation (29) and summing over the laminate thickness, for a laminate with n layers, $^1$ 

$$\begin{pmatrix}
N_{x} \\
N_{y} \\
N_{xy}
\end{pmatrix} = \sum_{k=1}^{n} \begin{bmatrix}
\overline{Q}_{11} & \overline{Q}_{12} & \overline{Q}_{13} \\
\overline{Q}_{12} & \overline{Q}_{22} & \overline{Q}_{23} \\
\overline{Q}_{13} & \overline{Q}_{23} & \overline{Q}_{33}
\end{bmatrix}_{k} \begin{cases}
\mathbf{J}_{z_{k-1}} \begin{pmatrix} \varepsilon_{x}^{0} \\ \varepsilon_{y}^{0} \\ \gamma_{xy}^{0} \end{pmatrix} dz + \mathbf{J}_{z_{k-1}} \begin{pmatrix} \kappa_{x} \\ \kappa_{y} \\ \kappa_{xy} \end{pmatrix} z dz \end{cases} (33)$$

$$\begin{pmatrix} M_{x} \\ M_{y} \\ M_{xy} \end{pmatrix} = \sum_{k=1}^{n} \begin{bmatrix} \overline{Q}_{11} & \overline{Q}_{12} & \overline{Q}_{13} \\ \overline{Q}_{12} & \overline{Q}_{22} & \overline{Q}_{23} \\ \overline{Q}_{13} & \overline{Q}_{23} & \overline{Q}_{33} \end{bmatrix}_{k} \begin{cases} \int_{z_{k-1}}^{z_{k}} \begin{pmatrix} \varepsilon_{x}^{0} \\ \varepsilon_{y}^{0} \\ \varepsilon_{y}^{0} \end{pmatrix} z dz + \int_{z_{k-1}}^{z_{k}} \begin{pmatrix} \kappa_{z} \\ \kappa_{y} \\ \kappa_{xy} \end{pmatrix} z^{2} dz \end{cases} (34)$$

 $[\epsilon^{\circ}]$  and  $[\kappa]$  are not functions of z, but are midplane values, so they can be removed from under the summation signs and Equations (33) and (34) can be written as  $^1$ 

$$\begin{pmatrix}
N_{x} \\
N_{y} \\
N_{xy}
\end{pmatrix} = \begin{bmatrix}
E_{11} & E_{12} & E_{13} \\
E_{12} & E_{22} & E_{23} \\
E_{13} & E_{23} & E_{33}
\end{bmatrix} \begin{pmatrix}
\varepsilon_{x}^{0} \\
\varepsilon_{y}^{0} \\
\gamma_{xy}^{0}
\end{pmatrix} + \begin{bmatrix}
B_{11} & B_{12} & B_{13} \\
B_{12} & B_{22} & B_{23} \\
B_{13} & B_{23} & B_{33}
\end{bmatrix} \begin{pmatrix}
\kappa_{x} \\
\kappa_{y} \\
\kappa_{xy}
\end{pmatrix} (35)$$

$$\begin{pmatrix} M_{x} \\ M_{y} \\ M_{xy} \end{pmatrix} = \begin{bmatrix} \overline{B}_{11} & B_{12} & B_{13} \\ B_{12} & B_{22} & B_{23} \\ B_{13} & B_{23} & B_{23} \end{bmatrix} \begin{pmatrix} \varepsilon_{x}^{0} \\ \varepsilon_{y}^{0} \\ \gamma_{xy}^{0} \end{pmatrix} + \begin{bmatrix} \overline{D}_{11} & D_{12} & D_{13} \\ D_{12} & D_{22} & D_{23} \\ D_{13} & D_{23} & D_{33} \end{bmatrix} \begin{pmatrix} \kappa_{x} \\ \kappa_{y} \\ \kappa_{xy} \end{pmatrix} (36)$$

where

$$E_{ij} = \sum_{k=1}^{n} [\overline{Q}_{ij}]_{k} (z_{k} - z_{k-1})$$
 (37)

$$B_{ij} = \frac{1}{2} \sum_{k=1}^{n} [\overline{Q}_{ij}]_{k} (z_{k}^{2} - z_{k-1}^{2})$$
 (38)

$$D_{ij} = \frac{1}{3} \sum_{k=1}^{n} \left[ \overline{Q}_{ij} \right]_{k} (z_{k}^{3} - z_{k-1}^{3})$$
 (39)

For laminates that are symmetric in both geometry and material properties about the middle surface, Equations (35) and (36) simplify considerably. In particular, because of the symmetry of  $[\overline{Q}_{ij}]_k$  and the lamina thickness  $t_k$ , all the  $[B_{ij}]$  are equal to zero and the force and moment resultants for a symmetric laminate are

$$\begin{pmatrix}
N_{x} \\
N_{y} \\
N_{xy}
\end{pmatrix} = \begin{bmatrix}
E_{11} & E_{12} & E_{13} \\
E_{12} & E_{22} & E_{23} \\
E_{13} & E_{23} & E_{33}
\end{bmatrix} \begin{pmatrix}
\varepsilon_{x}^{0} \\
\varepsilon_{y}^{0} \\
\gamma_{xy}^{0}
\end{pmatrix} (40)$$

For the remainder of this investigation, the laminate will be considered to be symmetrical and in a state of pure tension or compression, that is, bending moments will be zero and only Equation (40) will apply.

# Ply Failure Criteria

## Modified Maximum Strain Theory

Most experimental determinations of the strength of a material are based on uniaxial stress states. However, the practical problem usually involves at least a biaxial state of stress. For an orthotropic lamina, strength criteria parallel and transverse to the fiber direction due to tension, compression and shear strength may all be experimentally determined. To relate this uniaxial strength information to an analysis of ply damage and progressive failure, the following modified maximum strain theory is proposed. The lamina is said to have failed in the fiber direction if

$$\varepsilon_1 > X_{\varepsilon t}$$
 or  $\varepsilon_1 < X_{\varepsilon c}$  (42)

and transverse to the fibers if

$$\epsilon_2 > Y_{\epsilon t}$$
 or  $\epsilon_2 < Y_{\epsilon c}$  (43)

where  $X_{\varepsilon t}$ ,  $X_{\varepsilon c}$ , and  $Y_{\varepsilon t}$ ,  $Y_{\varepsilon c}$  indicate the maximum allowable tensile and compressive strains in the 1 and 2 directions. In the same way, the lamina is said to have failed in shear if

$$|\gamma_{12}| > S_{\varepsilon} \tag{44}$$

where  $\mathbf{S}_{_{\boldsymbol{\Xi}}}$  is the maximum allowable shear strain.

This failure theory makes it possible to obtain post-failure constitutive equations. However, stress or strain interactions, such as the combined effect of transverse strain and shear on failure, have been ignored.

## Post Failure Constitutive Equations

In forming the modified strain theory, the following assumptions were made:

- (1) If a lamina fails in the fiber direction, the matrix will still carry a load transverse to the fibers, but will not carry a shear load.
- (2) If a lamina fails transverse to the fiber direction, it will not carry a shear load, but the fibers will carry a normal load parallel to the fibers.
- (3) If a lamina fails in shear, the matrix will not carry a load transverse to the fibers, but the fibers will carry a normal load.
- (4) If a lamina fails in the fiber direction and in shear or fails both parallel and transverse to the fibers, the lamina is considered to have totally failed and will not support a load.

Each of the above assumptions indicates a partial or total failure of the lamina. Examination of Equation (4) shows that for a partial failure of a lamina at a given strain, the stress is changed by a change or softening of the stiffness matrix. In the computer solution of the biaxial stress problem, the changes in stiffness and load are used. Therefore, the post-failure lamina constitutive equations will be expressed in terms of change in stress and stiffness due to partial or total lamina failure.

Using Equation (4), the change in stress due to a change in stiffness is given by

$$\begin{pmatrix}
\Delta \sigma_{1} \\
\Delta \sigma_{2} \\
\Delta \tau_{12}
\end{pmatrix} = [\Delta Q_{ij}] \begin{pmatrix}
\varepsilon_{1} \\
\varepsilon_{2} \\
\gamma_{12}
\end{pmatrix}$$
(45)

where  $\,{\bf Q}_{\dot{1}\dot{\bf j}}$  is the change in stiffness due to failure, or

$$\Delta Q_{ij} = Q_{ij}$$
 - post failure stiffness (46)

The  $\Delta Q_{\mbox{\scriptsize ij}}$  terms for the various types of lamina failure are found as follows:

(1) Lamina failure in the fiber direction is assumed to cause  $E_{11}$ ,  $G_{12}$ ,  $v_{12}$ , to equal zero leaving  $E_{22}$  as the only factor contributing to the new  $Q_{ij}$  stiffness. Using Equations (5) and (46), the  $\Delta Q_{ij}$  terms are found to be

$$\Delta Q_{11} = \frac{E_{11}}{(1 - v_{12}v_{21})}$$

$$\Delta Q_{22} = \frac{E_{22}(v_{12}v_{21})}{(1 - v_{12}v_{21})}$$

$$\Delta Q_{12} = \frac{v_{12}E_{11}}{(1 - v_{12}v_{21})}$$

$$\Delta Q_{66} = G_{12}$$
(47)

(2) In the same way, lamina failure transverse to the fibers is assumed to cause  $E_{22}$ ,  $G_{12}$ ,  $v_{12}$  to equal zero leaving  $E_{11}$  as the only contributing stiffness factor. For this type failure the  $\Delta Q_{ij}$  terms are

$$\Delta Q_{11} = \frac{E_{11}(v_{12}v_{21})}{(1 - v_{12}v_{21})}$$

$$\Delta Q_{11} = \frac{E_{22}}{(1 - v_{12}v_{21})}$$

$$\Delta Q_{12} = \frac{v_{12}E_{11}}{(1 - v_{12}v_{21})}$$

$$\Delta Q_{66} = G_{12}$$
(48)

- (3) Lamina failure in shear is assumed to cause  $E_{22}$ ,  $G_{12}$ ,  $v_{12}$ , to equal zero leaving  $E_{11}$  as the only contributing stiffness factor. Thus, Equation (49) also gives the  $\Delta Q_{ij}$  terms for shear failure.
- (4) Lamina failure in the fiber direction and in shear, or failure both parallel and transverse to the fibers, is assumed to cause total lamina failure and therefore, zero remaining stiffness. The  $\Delta Q_{ij}$  terms obtained by Equations (5) and (46) are

$$\Delta Q_{11} = \frac{E_{11}}{(1 - v_{12}v_{21})}$$

$$\Delta Q_{22} = \frac{E_{22}}{(1 - v_{12}v_{21})}$$

$$\Delta Q_{12} = \frac{v_{12}E_{11}}{(1 - v_{12}v_{21})}$$

$$\Delta Q_{66} = G_{12}$$
(49)

Denoting  $\Delta Q_{ij}$  as the change in lamina stiffness with respect to the arbitrary xy axes and using Equations (8), (10), and (45), the change in stress in the xy coordinate system due to a change in stiffness is given by

$$\begin{pmatrix}
\Delta \sigma_{\mathbf{x}} \\
\Delta \sigma_{\mathbf{y}} \\
\Delta \tau_{\mathbf{x}\mathbf{y}}
\end{pmatrix}_{k} = \left[\Delta \overline{Q}_{\mathbf{i}\mathbf{j}}\right]_{k} \begin{pmatrix}
\varepsilon_{\mathbf{x}} \\
\varepsilon_{\mathbf{y}} \\
\gamma_{\mathbf{x}\mathbf{y}}
\end{pmatrix}_{k}$$
(50)

where

$$\left[\Delta \overline{Q}_{ij}\right]_{k} = \left[T_{ij}\right]_{k} \left[\Delta Q_{ij}\right]_{k} \left[T_{ij}\right]_{k}^{-1} \tag{51}$$

The  $\Delta Q_{ij}$  terms are calculated by use of Equations (17) and (18) where  $\Delta Q_{ij}$  terms are substituted for  $Q_{ij}$  terms.

#### Laminate Failure Criteria

With laminate strength, just as with the determination of laminate stiffness, the basic building block is the lamina with its inherent characteristics. Basic to determining the strength of a laminate is a knowledge of the stress state in each lamina. However, failure of one layer does not necessarily imply failure of the entire laminate. The laminate may, in fact, be capable of higher loads despite a significant change in stiffness.

The strength of an angle-ply laminate, symmetric about its middle surface, may be determined by examining the state of damage in each layer for a particular load. The laminate strains are calculated from the known load and stiffness prior to failure of a lamina. If one or more lamina have failed, as determined from the failure criterion, a new laminate stiffness is calculated and the laminate strains recalculated to determine the post-failure strains. Then it must be verified that the remaining laminae, at their increased strain levels, do not

fail at this applied load. Should an applied load cause progressive failure, where all layers successfully fail at the same load, the laminate is said to have suffered gross failure.

An alternative method, described in the next section, uses the original laminate stiffness to determine the strains at each load or failure cycle. When a failure takes place, a change in stiffness due to the failure is calculated. Using the change in stiffness and the known strains, a pseudo load is calculated and added to the original load, giving the required increase in strain. In an iterative finite element program this method is useful in that the stiffness matrix is only inverted once.

# Laminate Post-Failure Constitutive Equations

The strength of a symmetric angle-ply laminate subjected to plane stress is determined by first finding the strains for a known load. Inverting the stiffness matrix, Equation (40) can be written

$$\begin{cases}
\varepsilon_{x}^{o} \\
\varepsilon_{y}^{o} \\
\gamma_{xy}^{o}
\end{cases} = \left[E_{ij}\right]^{-1} \begin{Bmatrix} N_{x} \\ N_{y} \\ N_{xy} \end{Bmatrix}$$
(52)

From the previous assumption that plane sections perpendicular to the midplane axis remain plane, and for a state of plane stress, Equation (52) gives the state of strain for all layers. Then, by Equation (8), the strain with respect to the 1-2 axes for each layer k, is

$$\begin{cases}
\varepsilon_{1} \\
\varepsilon_{2} \\
\frac{\gamma_{12}}{2}
\end{cases}_{k} = \left[\mathsf{T}_{ij}\right]_{k} \begin{cases}
\varepsilon_{x}^{0} \\
\varepsilon_{y}^{0} \\
\frac{\gamma_{xy}}{2}
\end{cases}$$
(53)

The lamina strains are compared with the lamina failure criteria, Equations (42), (43), and (44), to determine modes of failure.

Should failures occur, changes in stiffness for each layer are calculated using Equations (47), (48), or (49), depending on the type of failure, and Equation (51). The total change in laminate stiffness is found by summing the laminar stiffness changes. For an n-ply laminate, the total change in stiffness,  $\Delta E_{ij}$ , is

$$[\Delta E_{ij}] = \sum_{k=1}^{n} [\Delta \overline{Q}_{ij}]_{k} (z_{k} - z_{k-1})$$
(54)

where  $\Delta \overline{Q}_{ij}$  is from Equation (51) and  $z_k - z_{k-1}$  is the thickness of lamina n.

Knowing the change in stiffness and the laminate strains, a pseudo force, PN, due to the loss of stiffness is found by

Adding this pseudo force to the applied load of Equation (52) gives the increased strain due to lamina failure. Then, the new strain is

Equations (53) through (56) are repeated until equilibrium is obtained or the laminate experiences gross failure.

# Application to Composite Structures

Thus far in this thesis, lamina and laminate have been analyzed in a state of plane stress, but the geometry and boundary conditions have not been considered. It has been assumed that the state of strain is constant throughout the laminate and that the stress is constant throughout each layer. In actual applications the stress in a laminate and in a lamina may vary considerably due to geometry and loading conditions. Stress concentrations such as holes, notches and cracks may increase the local stress to a much greater value than the stress at another point in the member. Such localized stresses may lead to localized laminate failure and ultimately to complete laminate failure at reduced loads.

In order to analyze a varying state of stress at points across a laminate, the finite element method will be used in conjunction with the previous ply and laminate equations.

#### NUMERICAL PROCEDURE

# Finite Element Method For Plane Stress Analysis

In a matrix analysis of composite materials the standard approach is to divide the composite laminate into a finite number of elements connected at joints or nodal points. The stiffness or flexibility properties of each individual element are then established by an element analysis, and the element stiffnesses combined to form the stiffness matrix for the complete structure. In the discussion that follows, a brief description of the displacement method for a constant strain triangle element will be presented and then incorporated with the previous ply and laminate equations in an iteration method to provide a solution to the nonlinear composite laminate problem.

Figure 4 depicts a typical triangular element with nodes i, j, and m, numbered in counter-clockwise order. Each node may have displacements in the x and y directions. Then, denoting displacements in the x and y directions by u and v respectively, the six components of element displacement may be written as the vector  $\{\delta\}^e$  where

$$\{\delta\}^{e} = \begin{cases} \begin{pmatrix} u_{i} \\ v_{i} \\ u_{j} \\ v_{j} \\ u_{m} \\ v_{m} \end{pmatrix}$$

$$(57)$$

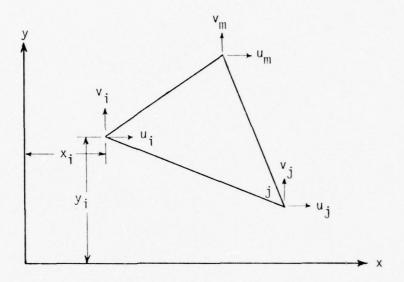


Figure 4. Plane Stress Triangular Element

The displacement within an element have to be uniquely defined by these six displacement values. Representing the displacements by two linear polynomials  $^{2\,3}$ 

$$u = \alpha_1 + \alpha_2 x + \alpha_3 y$$

$$v = \alpha_4 + \alpha_5 x + \alpha_6 y$$
(58)

the nodal displacements can be written

$$u_{i} = \alpha_{1} + \alpha_{2}x_{i} + \alpha_{3}y_{i}$$

$$u_{j} = \alpha_{1} + \alpha_{2}x_{j} + \alpha_{3}y_{j}$$

$$u_{m} = \alpha_{1} + \alpha_{2}x_{m} + \alpha_{3}y_{m}$$
(59)

$$v_{i} = \alpha_{4} + \alpha_{5}x_{i} + \alpha_{6}y_{i}$$

$$v_{j} = \alpha_{4} + \alpha_{5}x_{j} + \alpha_{6}y_{j}$$

$$v_{m} = \alpha_{4} + \alpha_{5}x_{m} + \alpha_{6}y_{m}$$

$$(60)$$

Evaluating the six constants  $\alpha$  in terms of the nodal displacements, gives  $^{2\,3}$ 

$$u = \frac{1}{2\Delta} (a_i + b_i x + c_i y) u_i + (a_j + b_j x + c_j y) u_j + (a_m + b_m x + c_m y) u_m$$
 (61)

and

$$v = \frac{1}{2} (a_i + b_i x + c_i y) v_i + (a_i + b_j x + c_j y) v_j + (a_m + b_m x + c_m y) v_m$$
 (62)

where

$$a_{i} = x_{j}y_{m} - x_{m}y_{i}$$

$$b_{i} = y_{j} - y_{m}$$

$$c_{i} = x_{m} - x_{j}$$
(63)

and

$$2\Delta = \det \begin{bmatrix} 1 & x_i & y_i \\ 1 & x_j & y_j \\ 1 & x_m & y_m \end{bmatrix} = 2 \text{ (area of triangle ijm)}$$
 (64)

Neglecting any initial strain, the total strain at any point within the element can be defined by its three components that contribute to the internal work. From Equations (24)

$$\begin{pmatrix}
\varepsilon_{X} \\
\varepsilon_{y} \\
\gamma_{xy}
\end{pmatrix} = \begin{pmatrix}
\frac{\partial u}{\partial x} \\
\frac{\partial v}{\partial y} \\
\frac{\partial u}{\partial y} + \frac{\partial v}{\partial x}
\end{pmatrix}$$
(65)

Using Equations (57), (61), (62), and (65), the strain within the triangular element expressed in terms of nodal displacement is

$$\begin{cases} \varepsilon_{\mathbf{x}} \\ \varepsilon_{\mathbf{y}} \end{cases} = [\mathbf{B}]^{\mathbf{e}} \{\delta\}^{\mathbf{e}}$$
 (66)

where

$$[B] = \frac{1}{2} \begin{bmatrix} b_{i} & 0 & b_{j} & 0 & b_{m} & 0 \\ 0 & c_{i} & 0 & c_{j} & 0 & c_{m} \\ c_{i} & b_{i} & c_{j} & b_{j} & c_{m} & b_{m} \end{bmatrix}$$
 (67)

and the B terms are

$$b_{i} = y_{i} - y_{m}$$
  $c_{i} = x_{m} - x_{j}$ 
 $b_{j} = y_{m} = y_{i}$   $c_{j} = x_{i} - x_{m}$ 
 $b_{m} = y_{i} - y_{j}$   $c_{m} = x_{j} - x_{i}$ 

(68)

By Equation (40), the stress resultant within an anisotropic element is

$$\begin{pmatrix}
N_{x} \\
N_{y} \\
N_{xy}
\end{pmatrix} = [E] \begin{pmatrix}
\varepsilon_{x} \\
\varepsilon_{y} \\
\gamma_{xy}
\end{pmatrix} = [E][B]^{e} \{\delta\}^{e}$$
(69)

Nodal forces can also be expressed in terms of the nodal displacements. Denoting U and V as the nodal forces in the x and y directions and assuming zero body forces, the nodal force vector  $\{F\}^e$  for a triangular element can be written

$$\{F\}^{e} = \begin{cases} U_{i} \\ V_{i} \\ U_{j} \\ V_{j} \\ U_{m} \\ V_{m} \end{cases}$$

$$(70)$$

The stresses that result from these nodal forces can be found by equating the work done by the forces to the strain energy stored in the element. The work done by the nodal forces is  $^{2\,3}$ 

$$W = \frac{1}{2} \left( U_{i} u_{i} + V_{i} v_{j} + U_{j} u_{j} + V_{j} v_{j} + U_{m} u_{m} + V_{m} v_{m} \right)$$
 (71)

or

$$W = \frac{1}{2} \left\{ F \right\}^{e^{\mathsf{T}}} \left\{ \delta \right\}^{e} \tag{72}$$

The strain energy is given by<sup>23</sup>

$$\overline{U} = \frac{t}{2} \int \int (\sigma_{x} \varepsilon_{x} + \sigma_{y} \varepsilon_{y} + \tau_{xy} \gamma_{xy}) dA$$

$$= \frac{t}{2} \{\sigma\}^{e^{T}} \{\varepsilon\}^{e} \Delta$$
(73)

where  $\Delta$  is the area of the triangular element and t is the element thickness. Using Equations (66) and (69), the strain energy can be

rewritten

$$\overline{U} = \frac{t\Delta}{2} \{\delta\}^{e^{\mathsf{T}}} [B]^{\mathsf{T}} [E]^{\mathsf{T}} [B] \{\delta\}^{e}$$
 (74)

Equating the work and energy equations and taking the transpose of both sides,

$$\{F\}^{e} = \Delta t [B]^{T}[E][B]\{\delta\}^{e}$$
(75)

Denoting the element stiffness matrix  $[K]^e$ ,

$$\{F\}^{e} = [K]^{e} \{\delta\}^{e} \tag{76}$$

and

$$[K]^{e} = \Delta t [B]^{\mathsf{T}}[E][B] \tag{77}$$

Equations (76) and (77) are now sufficient for computation with the actual matrix operations being accomplished in the computer program. Combining the element stiffness matrices and their force and displacement vectors gives the structural system of equations

$$[A] \{\delta\} = \{F\} \tag{78}$$

where [A] is the structural stiffness matrix.

# Solution Method for Nonlinear Material Properties

#### Initial Stress Process

The expressions derived in the previous sections describe fully the stress-strain relations for a laminated composite material in a state of plane stress. The essential nonlinearity is evident from

Equations (54), (55), and (56) with the composite stiffness matrix being dependent on the state of total stress. This problem, as described in the following section, can be approached using peacewise linearization to obtain a solution iteratively. 18,26

The "initial stress" process approaches the solution of a non-linear problem as a series of approximations.  $^{15^{-19},20^{-25}}$  In the first step after a load increment a purely elastic problem is solved determining an increment of strain  $\{\Delta \epsilon'\}$  and of stress  $\{\Delta \sigma'\}$  at every point of the structure. The nonlinearity implies that for the increment of strain found, the increment of stress will, in general, not be correct. If the true increment of stress for equilibrium is  $\{\Delta \sigma\}$ , then the correct solution can be maintained by a set of pseudo body forces equilibrating the "initial stress" system  $\{\Delta \sigma'\}$  -  $\{\Delta \sigma\}$ .  $^{19}$ 

At the second stage of computation the system of pseudo body forces can be removed by allowing the structure (with unchanged elastic properties) to deform further. An additional set of strain and stress increments is caused, and once again they are likely to exceed those permitted by the nonlinear problem. The redistribution of pseudo body forces is repeated and the process continued until it converges to the nonlinear equilibrium conditions.

# Application to Composite Materials

In laminated composite materials, the nonlinearity comes from failure or partial failure of a ply within a laminate. Ply failure or partial failure implies that a change in stiffness has taken place and that the load used and displacements found, for an elastic solution, are not correct. To arrive at the correct solution, pseudo body

forces are calculated using the change in stiffness and the laminate strains. These pseudo forces are allowed to further deform the laminate using the original elastic properties. New strains are found and the process repeated until equilibrium is obtained.

Specific steps in the initial stress process as applied to composite materials are:

(1) The problem is set up by using Equation (77), for each element, to construct the structural stiffness matrix. An incremental load and other boundary conditions are entered into Equation (73) which gives the following system of equations to be solved:

$$[A] \{\delta\} = \{F\} \tag{79}$$

- (2) The stiffness matrix [A] is partially inverted and the displacement  $\{\delta\}$  computed.
- (3) Strains within each element are found by Equation (66)

$$\begin{cases} \varepsilon_{x} \\ \varepsilon_{y} \\ \gamma_{xy} \end{cases} = [B]^{e} \{\delta\}^{e}$$
 (80)

(4) From Equation (8), the principal strains in each lamina are obtained,

$$\begin{pmatrix}
\varepsilon_{1} \\
\varepsilon_{2} \\
\frac{\gamma_{12}}{2}
\end{pmatrix}_{k}^{e} = \left[T\right]_{\kappa} \begin{pmatrix}
\varepsilon_{x} \\
\varepsilon_{y} \\
\frac{\gamma_{xy}}{2}
\end{pmatrix}$$
(81)

- (5) Lamina failure criteria, Equations (42), (43), and (44) are applied. If there are no failures, go to step 10.
- (6) The change in element stiffness due to failure is computed using Equation (54),

$$[\Delta E] = \sum_{k=1}^{n} [\Delta Q]_{\kappa} (z_k - z_{k-1})$$
(82)

(7) Pseudo forces at each node point are found using [ $\Delta E$ ] as the element stiffness and Equations (66) and (75). Denoting the pseudo forces PF,

$$\{PF\} = [B]^{\mathsf{T}}[\Delta E] \begin{cases} \varepsilon_{\mathsf{X}} \\ \varepsilon_{\mathsf{y}} \\ \gamma_{\mathsf{X}\mathsf{y}} \end{cases} \Delta \mathsf{t}$$
 (83)

- (8) The maximum pseudo force,  $PF_{max}$ , is compared with an accuracy constant ACC. If  $PF_{max}$  is less than ACC, equilibrium has been reached, go to step 10.
- (9) Using the pseudo forces and the partially inverted stiffness matrix of step 2, additional displacements are found and added to the original displacements. Return to step 3.
- (10) The load is incremented by adding the displacements obtained for an incremental load to existing displacements and returning to step 3.

Should the iteration process of steps 3 through 9 be repeated 20 times within an increment without reaching equilibrium, the laminate is considered to have suffered gross failure and the process is stopped.

#### SOLUTION OF PROBLEMS

#### Uniaxial Tension Specimens

To check the reliability of the modified strain theory and the finite element program, the predicted stress-strain curves and failure loads for  $(0^{\circ})_s$ ,  $(90^{\circ})_s$ ,  $(0^{\circ}/90^{\circ}/90^{\circ}/0^{\circ})_s$ ,  $(90^{\circ}/\pm 45^{\circ}/0^{\circ})_s$  and  $(90^{\circ}/\pm 45^{\circ}/90^{\circ})_s$  laminates were obtained for uniaxial tension loads. Figure 5 shows the finite element grid used. The 12 element grid, scaled to 5 inches (12.7 cm) in length and 1 inch (2.54 cm) in width, was loaded using incremental displacements in the direction shown. Zero displacement conditions were specified for nodes opposite the load end in the load direction and along one side transverse to the load direction.

To establish a stress-strain relationship for comparison to experimental data, one element was chosen and its state of stress and strain written out at the end of each increment. The failure status of eacy ply within each element, nodal displacements, iterations and the maximum pseudo force for each iteration were also written out.

Material properties used were those for Thornel 300/5208 graphite-epoxy, listed in Appendix A.

#### Circular Hole Specimens

Two laminates containing circular holes and loaded in uniaxial tension were investigated using the finite element grid shown in

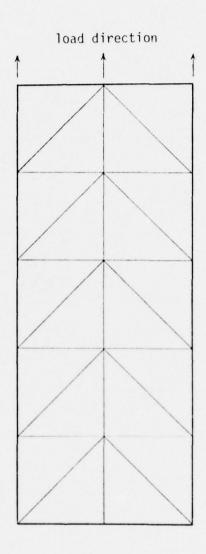


Figure 5. Finite element grid, uniaxial tension test

Figure 6. The 99 element grid was given an incremental displacement load in the direction shown, with zero displacement conditions imposed on the end opposite the load in the load direction and along the hole side transverse to the load direction. The scale for the grid represented a specimen 3 inches (7.62 cm) wide with a hole 1 inch (2.54 cm) in diameter.

At the end of each increment the status of eacy ply within each element was printed out. Total failure loads were determined using the nodal displacements and stiffnesses at failure to calculate the nodal forces at the load points.

Material properties used were those for Thornel 300/5208 graphite-epoxy, listed in Appendix A.

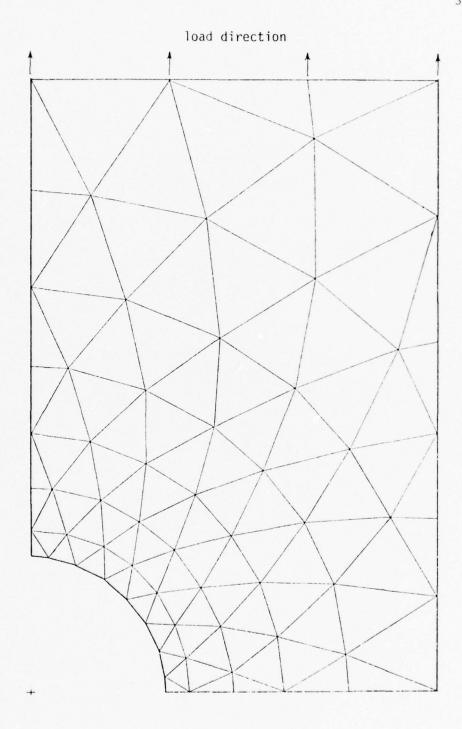


Figure 6. Finite element grid, circular hole in uniaxial tension

#### RESULTS

# Uniaxial Tension Specimens

Stress vs. strain diagrams for the uniaxial tension problems are given in Figures 7 through 11. Data points plotted are those obtained by Sendeckyj for Thornell 300/5208 graphite-epoxy laminates<sup>2,9</sup> The solid line is the predicted stress-strain curve as obtained from a chosen element. Changes in nodal displacements written out at the end of each increment were equal for all elements, indicating the strains, and thus the stresses, were equal for all elements.

# Circular Hole Specimens

# $(0^{\circ}/90^{\circ}/90^{\circ}/0^{\circ})_{s}$ Laminate

Figures 12 through 16 give the damage or progressive failure status of the  $(0^\circ/90^\circ/90^\circ/0^\circ)_s$  laminate at the end of each load increment. The first number code indicates the status of the  $0^\circ$  plies and the second number the status of the  $90^\circ$  plies. The specific numbers give the mode of failure within the ply.

The total failure load calculated for this notched laminate was 37,600 psi (2.6 x  $10^8$  Pa) while that obtained experimentally by Nuismer and Whitney was 28,200 psi (1.9 x  $10^8$  Pa).<sup>30</sup>

~ 30% high

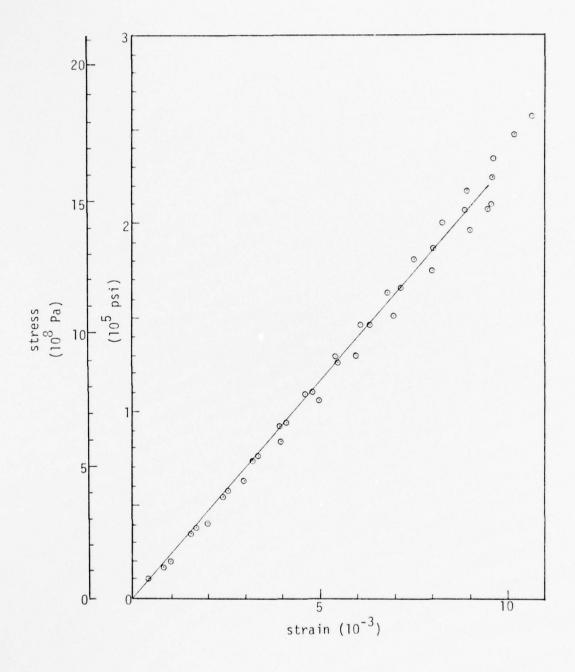


Figure 7. Stress vs strain,  $\left(0^{\circ}\right)_{S}$  laminate in uniaxial tension

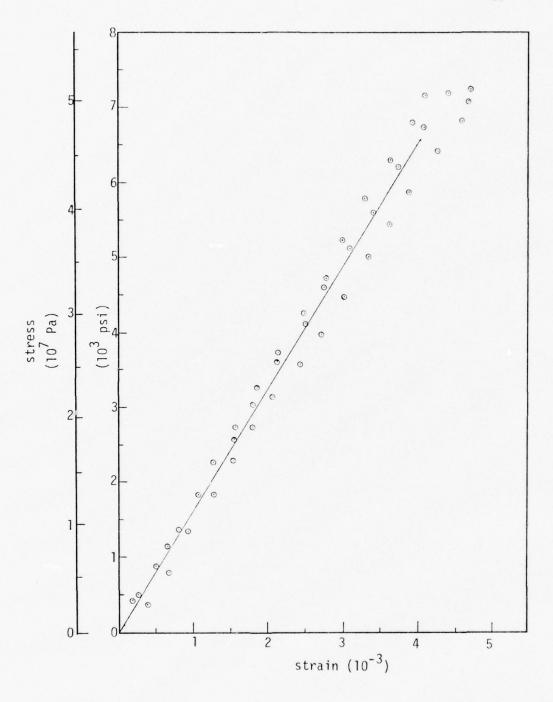


Figure 8. Stress vs strain,  $(90^{\circ})_{\rm S}$  laminate in uniaxial tension

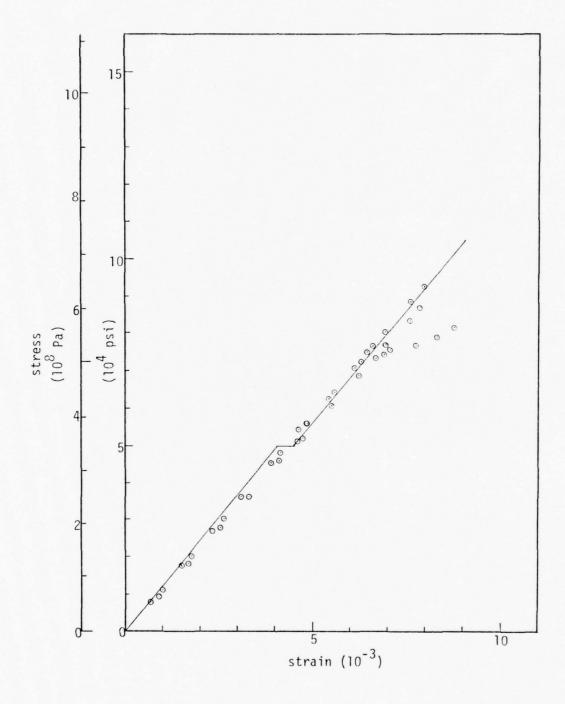


Figure 9. Stress vs strain,  $(0^{\circ}/90^{\circ}/90^{\circ}/0^{\circ})_{S}$  laminate in uniaxial tension

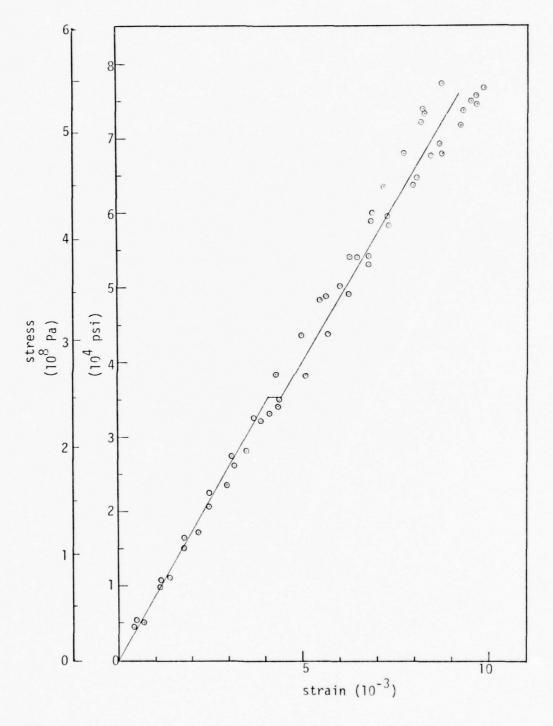


Figure 10. Stress vs strain,  $(90^{\circ}/\pm 45^{\circ}/0^{\circ})_{s}$  laminate in uniaxial tension

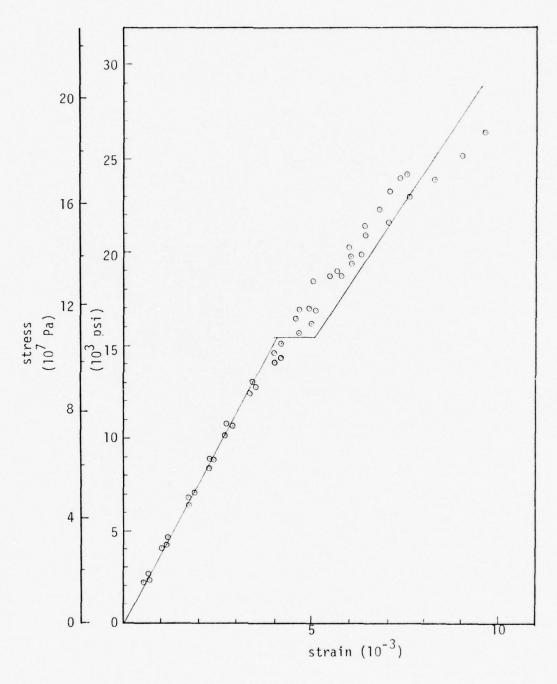


Figure 11. Stress vs strain  $(90^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$  laminate in uniaxial tension

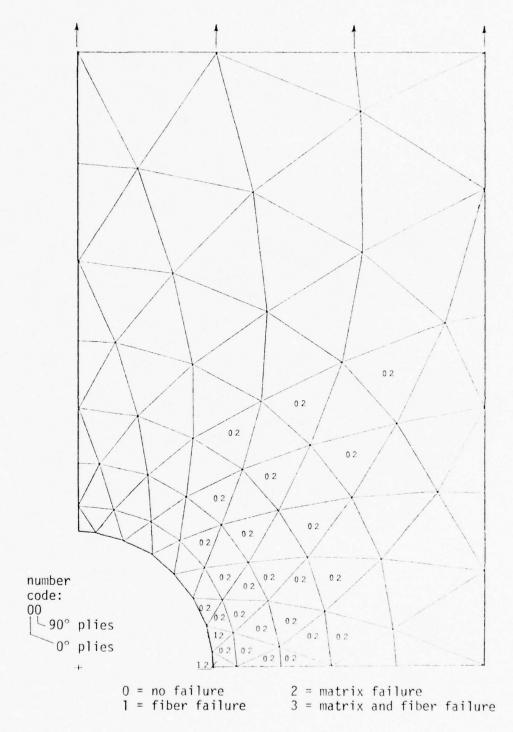


Figure 12.  $(0^\circ/90^\circ/90^\circ/0^\circ)_s$  circular hole partial failure at .009 in. (.023 cm) displacement load

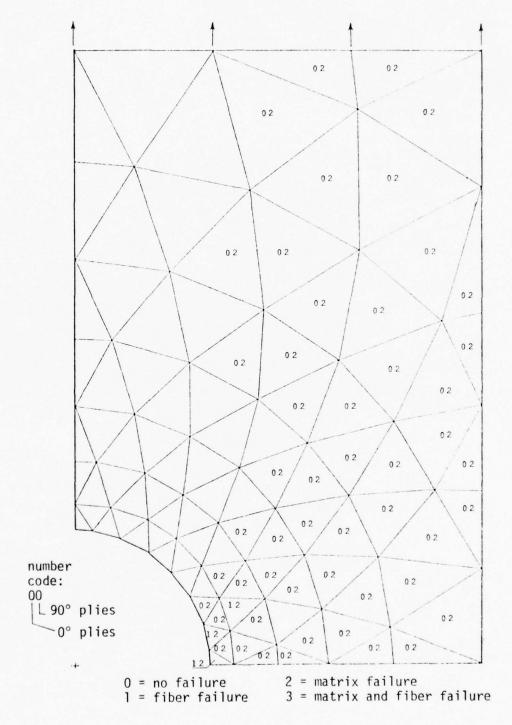


Figure 13.  $(0^{\circ}/90^{\circ}/90^{\circ})_{s}$  circular hole partial failure at .012 in. (.030 cm) displacement load

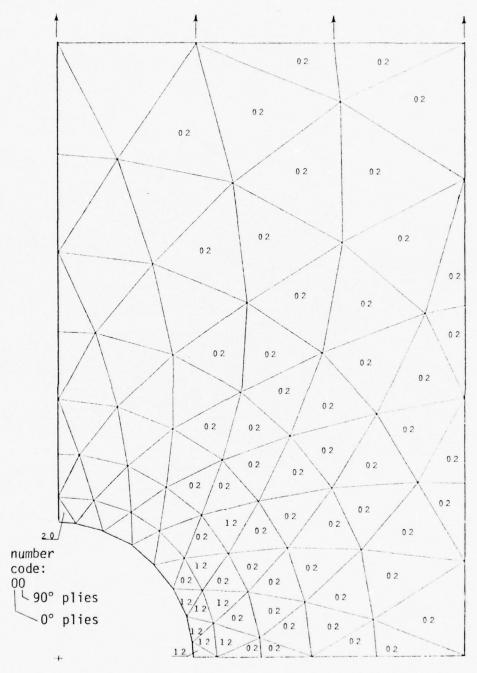


Figure 14.  $(0^{\circ}/90^{\circ}/90^{\circ})_{\rm S}$  circular hole partial failure at .015 in. (.038 cm) displacement load

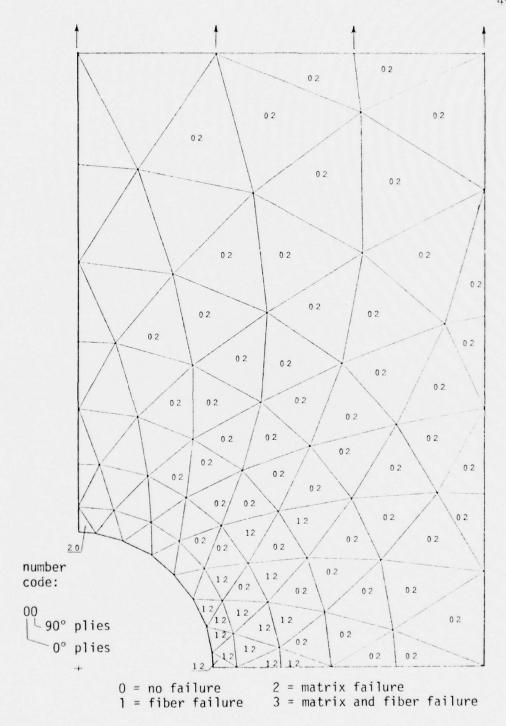


Figure 15.  $(0^{\circ}/90^{\circ}/90^{\circ}/0^{\circ})_{s}$  circular hole partial failure at .018 in. (.046 cm) displacement load

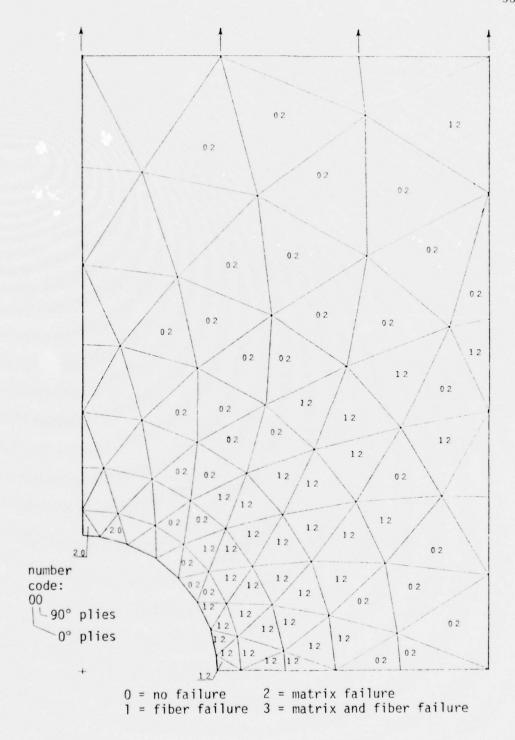


Figure 16.  $(0^{\circ}/90^{\circ}/90^{\circ}/0^{\circ})_{s}$  circular hole, complete failure at .021 in. (.053 cm) displacement load

(1/4)5

# $\left(0^{\circ}/\pm\ 45^{\circ}/90^{\circ}\right)_{\text{S}}$ Laminate

Figures 17 through 20 give the damage or progressive failure status of the  $(0^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$  laminate at the end of each load increment. The numbers in the code indicate the failure status of the plies as follows: the first number for the  $0^{\circ}$  ply, the second number for the  $+ 45^{\circ}$  ply, the third number for the  $- 45^{\circ}$  ply and the fourth number for the  $90^{\circ}$  ply.

The total failure load was calculated to be 47,800 psi  $(3.3 \times 10^8 \text{ Pa})$  while an experimental value of 45,700 psi  $(3.2 \times 10^8 \text{ Pa})$  was obtained by Nuismer and Whitney.  $^{3.0}$ 

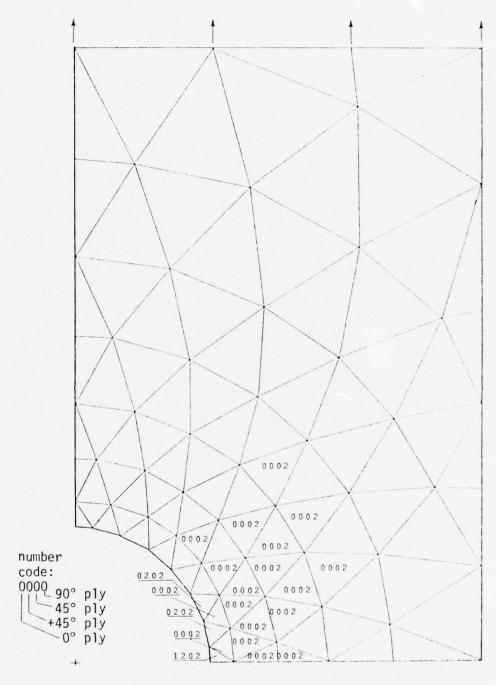


Figure 17.  $(0^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$  circular hole partial failure at .008 in. (.020 cm) displacement load

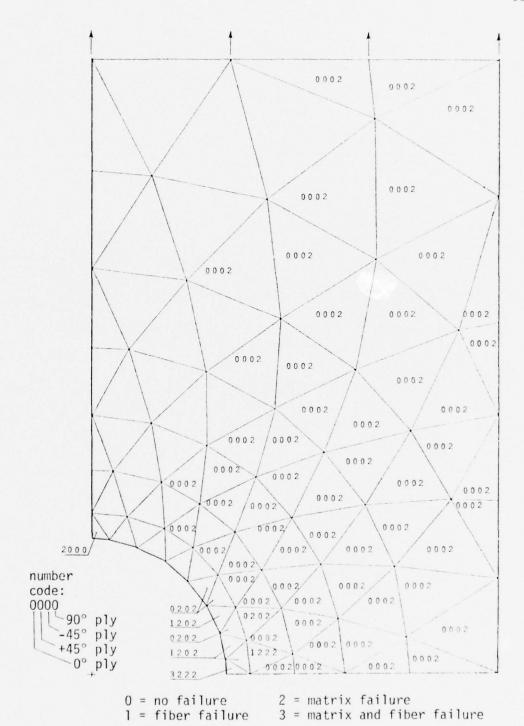
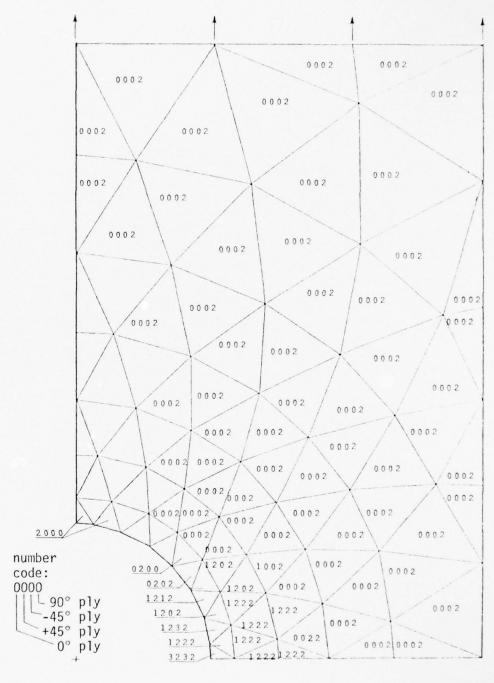


Figure 18.  $(0^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$  circular hole partial failure at .012 in. (.030 cm) displacement load



0 = no failure 2 = matrix failure 1 = fiber failure 3 = matrix and fiber failure

Figure 19.  $(0^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$  circular hole partial failure at .016 in. (.041 cm) displacement load

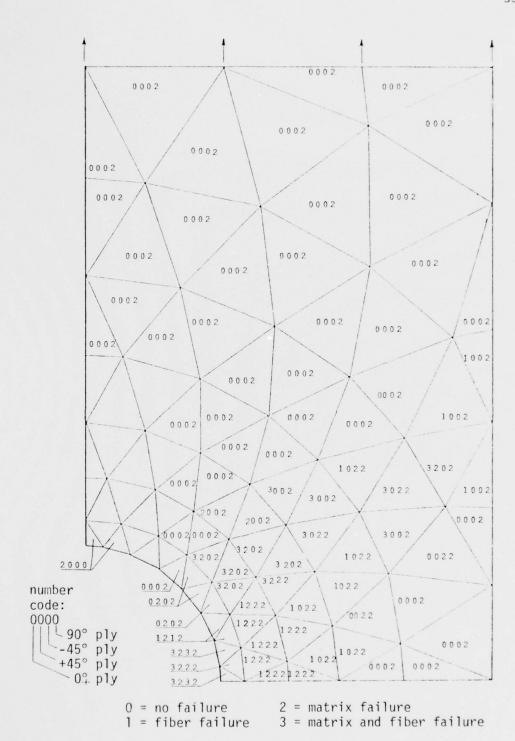


Figure 20.  $(0^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$  circular hole, complete failure at .020 in. (.051 cm) displacement load

#### DISCUSSION

In general, the theoretical stress-strain results, Figures 7 through 11, for the unnotched tensile specimens compare favorably with experimental data with some variation in the ultimate failure point of the laminate. Figures 7 and 8 show the theoretical failure stresses and strains for the  $0^{\circ}$  and  $90^{\circ}$  laminates to be somewhat below the experimental failure values. These differences might be the result of choosing low values for the maximum allowable strains. However, Figures 9 and 11, using the same maximum allowable strains, indicate theoretical ultimate strengths for the  $(0^{\circ}/90^{\circ}/90^{\circ}/0^{\circ})_{s}$  and  $(90^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$  laminates slightly higher than the experimental values. The low experimental failure values of Figure 9 were suspected to be due to damage to the surface 0° plies during specimen handling and fabrication. 29 In Figure 11, where the + 45° plies are the dominant load carrying plies, deviation from the experimental values may be influenced by the assumption that the shear stress vs strain is linear, when in actuality, it is highly nonlinear. 29  $(90^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$  theoretical results, Figure 10, compare well with experimental data. The horizontal jumps in the theoretical curves of Figures 9, 10, and 11, indicate increased strain due to failure of the 90° plies.

The progressive failure of the circular hole specimens produced interesting results in that the failure of both specimens began at

the hole edge in a direction perpendicular to the load direction but did not progress in the shortest direction to the specimen outer edge. Figures 12 through 16, and Figures 17 through 20, seem to indicate that for both the  $(0^{\circ}/90^{\circ}/90^{\circ}/0^{\circ})_{s}$  and  $(0^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$  laminates the failure path was approximately 45 above the shortest, or horizontal path across the grid. However, it should be noted that although the failure status of the elements in Figures 16 and 20 probably give a good indication of the ultimate failure modes, they may not be exact because equilibrium of the pseudo nodal forces was not obtained for the failure load.

Figures 12 through 16 show the 90° plies of the  $(0^\circ/90^\circ/90^\circ/0^\circ)_s$  laminate failing first in the matrix as expected, with the 0° plies following essentially the same element failure pattern at higher loads. The ultimate failure load of 37,600 psi  $(2.6 \times 10^8 \text{ Pa})$  obtained theoretically was much higher than the 28,200 psi  $(1.9 \times 10^8 \text{ Pa})$  experimental value. This error of over 30 percent possibly indicates that the modified strain theory used needs refinement.

Figures 17 through 20, for the  $(0^{\circ}/\pm 45^{\circ}/90^{\circ})_{s}$  laminate show the 90° ply failing first in the matrix, with this failure progressing throughout most of the structure before the ultimate failure load was reached. Failure of the  $\pm$  45° and 0° plies is more limited throughout the load range and indicates the progressive path of total failure. Although the theoretical failure load of 47,800 psi (3.3 x  $10^{8}$  Pa) is close to the experimental value of 45,700 psi (3.2 x  $10^{8}$  Pa), this experimental value is suspected to be too high. The error is suspected because, in the study by Nuismer and Whitney, 30 other failure

stresses decreased with increasing hole size as expected, whereas this particular value increased. Comparison with other data in this study<sup>30</sup> indicates that the actual failure stress might be around  $40,000 \text{ psi} (2.8 \times 10^8 \text{ Pa})$ . If this is the case, the error is considerable and perhaps is again pointing to a necessary refinement of the modified strain theory.

In all problem solutions, it was assumed that the lamina had the same stress-strain curve in compression as tension. Also, compressive-failure strains were assumed to be the same as those for tension. These assumptions, made partially because of the lack of reliable compression data, are certainly incorrect and would have to be modified for problems involving substantial compression. However, for the hole problems considered, where compression occurs only in a small region at the top of the hole, this is not expected to result in the appreciable errors.

One other factor not considered in the investigation is the free edge effects. The assumption was made that the strain through the laminate was constant at any given point. With this assumption, the stresses in plies at different orientations will generally be different. At free edges, although the average stress along the free edge is set equal to zero, this leads to mathematically non-zero surface tractions along the free edge of each ply, thus violating the actual boundary conditions. If the actual boundary conditions are used, it can be shown to result in significant interlaminar shear and normal stresses that have been shown to be responsible for delamination along free boundaries. However, this effect has been shown to perturb

the inplane stresses predicted from laminated plate theory in only a small region near the boundary. Furthermore, observation of notched specimens subjected to monotonic failure loads has produced no evidence of delamination at the notch before failure occurs. Thus, the free edge effect is expected to be of importance in the progressive failure of notched laminates only if fatigue loadings are considered.

#### APPENDIX A

#### THORNEL 300/5208 GRAPHITE-EPOXY PROPERTIES

$$E_{11} = 23 \times 10^6 \text{ psi} \quad (15.9 \times 10^{10} \text{ Pa})$$

$$E_{12} = 1.6 \times 10^6 \text{ psi} (1.1 \times 10^{10} \text{ Pa})$$

$$G_{12} = .77 \times 10^6 \text{ psi} (.53 \times 10^{10} \text{ Pa})$$

$$X_{et} = 9.5 \times 10^{-3}$$

$$X_{\varepsilon c} = 9.5 \times 10^{-3}$$

$$Y_{et} = 4.1 \times 10^{-3}$$

$$Y_{\varepsilon C} = 4.1 \times 10^{-3}$$

$$S_{\varepsilon} = 23 \times 10^{-3}$$

#### APPENDIX B

#### COMPUTER PROGRAM DESCRIPTION

#### Main Program

The Main program is the executive routine that controls the sequence of steps by calling subroutines that set up and execute the problem. Subroutines called by Main are given in the Computer Program section.

#### Subroutines

#### Setup

Subroutine Setup is called by the Main program. It reads the data deck, adjusts the structure size by scaling factors, checks the bandwidth and adjusts it if necessary. Setup then calls Stifgn, and upon return of control, writes out the data deck and returns control to Main.

#### Stifgn

Subroutine Stifgn computes the laminate stiffness matrix by using the orthotropic lamina properties and Equations (17), (18), and (37). Control is returned to Setup.

#### Constr

Subroutine Constr. is called by Main. It controls the construction of the system stiffness matrix by calling Subroutine Elcons for each element, and then returns control to Main.

#### Elcons

Subroutine Elcons is called by Constr. This subroutine constructs the stiffness matrixes for the individual elements by using Equation (77), and then combines the element stiffnesses to form the structure stiffness matrix. Control is returned to Constr.

#### Excite

Subroutine Excite is called by the Main program and enters the problem boundary conditions. If the boundary conditions are specified forces, these forces are entered directly into the force matrix. For displacement boundary conditions, a pseudo force dependent only on the specified displacement is entered into the force matrix, the corresponding diagonal elements of the stiffness matrix are not changed, but assumed equal to one, and the remaining terms in the related rows and columns are set equal to zero. Control is returned to Main.

#### Gausel

Subroutine Gausel is called by the Main program. It solves the boundary value problem by Gaussian elimination, leaving the stiffness matrix in partially inverted form. This partially inverted matrix is subsequently used in Subroutine Foredu during iteration. Control is returned to Main.

#### Elfail

Subroutine Elfail is called by the Main program. This subroutine controls the system failure analysis, iteration, incrementation and output. Subroutines called by Elfail are: Strain, Layer,
Pforce, Foredu, and Output. Upon failure to reach equilibrium
within 20 iterations, control is returned to Main.

#### Strain

Subroutine Strain is called by Elfail. It calculates element strains using Equation (66) and returns control to Elfail.

#### Layer

Subroutine Layer is called by Elfail. This subroutine applies the failure criteria to each lamina within each element. Should failure occur, it is noted in the failure tracing array ITT, Subroutine Dlstif is called, then control is returned to Elfail. If no failures occur, control is returned to Elfail.

#### Distif

Subroutine Distif is called by Layer. It calculates the change in element stiffness due to failures determined in Layer. Calculations are made using Equation (54) and control returned to Layer.

#### Pforce

Subroutine Pforce is called by Elfail. Using Equation (83), this subroutine calculates the pseudo nodal forces due to changes in stiffness. Control is returned to Elfail.

#### Foredu

Subroutine Foredu is called by Elfail. It uses the partially inverted stiffness matrix to reduce the force matrix and then back substitutes to solve for displacements due to iteration or incrementation. These displacements are added to the total displacement vector and control is returned to Elfail.

#### Output

Subroutine Output is called by Elfail. At the end of each increment or upon failure to reach equilibrium, Output writes out the applied displacement or load, the failure status of each lamina within each element, the nodal displacements and the pseudo nodal forces. Control is returned to Elfail.

#### APPENDIX C

#### COMPUTER PROGRAM VARIABLES

A	System stiffness matrix
F0	Force matrix, nodal values
FIN	Initial force matrix with boundary conditions entered, used in incrementation $ \\$
DIS	Delta displacement vector, nodal values
TDIS	Total displacement vector, nodal values
Q	Orthotropic lamina stiffness
EE	Delta stiffness due to failure
E11	Ell through E33 make up the laminate stiffness matrix
E13	
E22	
E23	
E33	
ANG2R	Two times lamina angle in radians
THETA	Lamina angle in radians
ITT	Failure tracing matrix
STRN	Strain in element chosen for output
STRS	Stress in element chosen for output
F00	Force matrix use in iteration
FOMAX	Maximum nodal force during iteration

EPX	Element strain in the x direction
EPY	Element strain in the y direction
GXY	Element shear strain
ESTRN	Lamina strain matrix referenced to xy coordinate system
LMSTRN	Lamina strain matrix referenced to the principal lamina axes
LMSTRS	Lamina stress matrix referenced to the principal lamina axes
В	B transpose used in Subroutine Pforce
BB	Calculation matrix used in Subroutine Pforce

Calculation matrix used in Subroutine Pforce

EP

#### APPENDIX D

#### COMPUTER PROGRAM INPUT DATA

IBD = Bandwidth; 2(largest difference in node numbers + 1)

NRD = Matrix order; 2(number of nodes)

NEL = Number of elements

NXY = 1; Coordinate data is input sequentially

= 2; Coordinate data is input as one pair per card

NM = Nodal numbers for each element in counter-clockwise order

XYM = Nodal coordinates, ordered pairs, x-coordinate first

SFX = X-scaling factor

SFY = Y-scaling factor

NXD = Number of non-zero applied displacements, x-face

NXF = Number of non-zero applied forces, x-face

NX1, 2 = End points of integration path to get a total force, x-face

NSX = 1; Resulting force is based on loads applied to the x-face
= 2; Resulting force is based on displacements applied to
the x-face

NYD = Number of non-zero applied displacements y-face

NYF = Number of non-zero applied forces, y-face

NY1, 2 = End points of integration path to get a total force, y-face

NZC = Total number of zero displacements

NANG = Number of unique ply orientations

NFAIL = 1; Maximum strain failure = 2; Maximum stress failure

NDPX = Array positions of coordinate numbers of non-zero applied displacements, x-face (2 X node number -1)

NFPX = Array positions of coordinate numbers of non-zero applied forces, x-face (2 X node number -1)

EX (1) = Magnitude of applied displacement increment, x-face

EX (2) = Magnitude of applied force increment, x-face

NX = Nodal numbers adjacent to applied force nodes, x-face

NDPY = Array positions of coordinate numbers of non-zero applied displacements, y-face (2 Y node number)

NFPY = Array positions of coordinate numbers of non-zero applied forces, y-face (2 X node number)

EY (1) = Magnitude of applied displacement increment, y-face

EY (2) = Magnitude of applied force increment, y-face

NY = Nodal numbers adjacent to applied force nodes, y-face

NZP = Array identification numbers for zero displacement conditions

El = Orthotropic material modulus in fiber direction

E2 = Orthotropic material modulus transverse to fibers

G = Orthotropic material shear modulus

V12 = Orthotropic material major Poisson's ratio

ANGLE = Orientation angles of individual plies, positive counterclockwise from the x-axis, in degrees

THK = Thickness of all plies at each unique orientation

ALLOW(1,1) = Limiting ply tensile strain, parallel to fibers

ALLOW(2,1)= Limiting ply tensile strain, transverse to fibers

ALLOW(3,1) = Limiting ply shear strain

ALLOW(4,1)= Limiting ply tensile stress, parallel to fibers

ALLOW(5,1) = Limiting ply tensile stress, transverse to fibers

ALLOW(6,1)= Limiting ply shear stress

ALLOW(1,2)= Limiting ply compressive strain, parallel to fibers

ALLOW(2,2)= Limiting ply compressive strain, transverse to fibers

ALLOW(3,2)= Limiting ply shear strain

ALLOW(4,2) = Limiting ply compressive stress, parallel to fibers

ALLOW(5,2)= Limiting ply compressive stress, transverse to fibers

ALLOW(6,2)= Limiting ply shear stress

ACC = Iteration accuracy factor

NEM = Element number of element chosen for stress and strain outputs

#### APPENDIX E

#### COMPUTER PROGRAM

#### Main

MAIN
CALL SETUP
CALL CONSTR
CALL EXCITE
CALL GAUSEL
CALL ELFAIL
STOP
END

#### Subroutine Setup

```
SUBROUTINE SETUP
   COMMON / ARRAYS / A(30,200), FO(200), FIN(200), DIS(200), TDIS(200)
  COMMON / CONSTI / MES, MSX, MSY, MANG, NFAIL, MXI, MX2, MY1, MY2
  COMMON / GEOMET / 180, NEL, LRD, NZC, NM(450), NZP(40), XYM(200)
  COMMON / MATUAT / E11, E12, E13, E22, E23, E33, G(3,3), EE(3,3)
  COMMON / XLOADS / NXU, NXF, NOPA(10), NFPX(10), NX(2,10), EX(2)
   COMMON / YLOADS / NYD, NYF, NDPY(10), NFPY(10), NY(2,10), ET(2)
   COMMON / ANISOT / E1, E2, G, V12, THK(5), ANGLE(5), ALLOH(6,2), HT
  COMMON / COMPT1 / ANG2R(5), THETA(5), ITT(150,5)
  COMMON / COMPT2 / ACC, NEM
   KEAD (5,1010) IBU, NRD, NEL, NXY
  NE3 = NEL + 3
  REAU (5,1020) (NM(I), I = 1, NE3)
   GU TO (10, 20), NXY
10 REAU (5,1030) (XYM(I), I = 1,NRD)
  60 TO 40
20 UO 30 J = 2,11RD,2
    I = J - 1
    READ (5,1030) XYM(I), XYM(J)
30 CONTINUE
40 KEAD (5,1040) SFX, SFY
   00 50 J = 2,NRD,2
    I = J - 1
    XYM(I) = XYM(I) * SFX
    XYM(J) = XYM(J) * SFY
50 CONTINUE
   00 60 I = 1.NEL
    13 = 1 * 3
   N1 = NA(13-2)
   N2 = NM(13-1)
   N3 = 11M(13)
   11 = MAXU(N1, N2, N3)
    JJ = MINO(N1, N2, N3)
   IJ = (II - JJ) * 2 + 2
```

```
IF (IJ .LE. IED) 60 10 60
     WRITE (0,5000) 1, 160, 1J
     100 = 13
  60 CONTINUE
    KEAD (5,1010) HXU, NXF, NXI, NX2, NXX, NYD, NYF, NY1, NY2, 1.SY,
                   NZC, NANG, NFAIL
     IF (NXD .EQ. 0) 60 TO 70
     KEAU (5,1010) (HUPX(1), 1 = 1, NXD)
     READ (5,1050) EX(1)
  70 IF (NXF .LG. 0) 60 TO 80
    READ (5,1010) (HEPX(1), 1 = 1,NXF)
     REAU (5,1060) EX(2)
    REAU (5,1010) ((kx(I,J), I = 1,2), J = 1,NXF)
  30 IF (HYD .EG. 0) 60 TO 90
    READ (5,1010) (NUPY(1), I = 1,NYD)
     KEAU (5,1050) FY(1)
  96 IF THIF .EG. 01 60 TO 100
     READ (5,1010) (HEFY(1), 1 = 1,NYF)
     HEAU (5,1060) EY(2)
    REAU (5,1010) ((RY(I,J)) I = 1,2), J = 1,NYF)
 100 REAU (5,1010) (1,2P(1), 1 = 1,11ZC)
    NEAU (5,1040) E1, E2, G, V12
     MEAD (5,1070) (ANGLE(I), I = 1, MANG)
     READ (5,1060) (THK(I), 1 = 1, NANG)
    REAU (5,1090) ((ALLOW(I,J), I = 1,6), J = 1,2)
    REAU (5,1080) ACC
    KEAU (5,1010) NEM
     CALL STIFGN
     MRITE (6,5010)
     WRITE (6,5030) IBD, NAD, NEL, NXY
     WRITE (6,5040) (NH(I), I = 1,NE3)
     WRITE (6,5050) (XYM(I), I = 1,NRD)
     WRITE (0,5000) SFX, SFY
     WRITE (0,5030) HAD, NXF, HX1, NX2, NSX, NYD, NYF, NY1, NY2, HSY,
                   NZC. NANG. NEATL
     IF (NXD .EQ. 0) GO TO 110
     WRITE (0,5030) (NUPX(I), I = 1,NXD)
     WRITE (6,5070) EX(1)
 110 IF (NXF .EU. 0) 60 TO 120
     mRITE (6,5030) (NFPX(1), I = 1,NXF)
     WRITE (0,5080) EX(2)
     WRITE (6,5030) ((NX(I,J), I = 1,2), J = 1,NXF)
 120 IF (NYD .EQ. 0) GO TO 130
     "RITE (6,5030) (NUPY(1), I = 1,11YD)
     WRITE (6,5070) EY(1)
 130 IF (NYF .EO, 0) GO TO 140
     WRITE (0,5030) (NFPY(1), I = 1,NYF)
     WRITE (0,5000) EY(2)
     WHITE (6,5030) ((NY(I,J), I = 1,2), J = 1,NYF)
 140 WRITE (6,5030) (NZP(I), I = 1,NZC)
     WRITE (6,5000) E1, E2, G, V12
     WRITE (6,5090) (ANGLE(I), I = 1, NANG)
     WRITE (6,0000) (THK(I), 1 = 1, NANG)
     WHITE (0,6010) ((ALLOW(I,J), J = 1,2), I = 1,6)
     ARITE (0,5070) ACC
     WRITE (6,5030) NEM
     #KITE (6,6020)
     KETURN
1010 FORMAT (20(13,1x))
1020 FORMAT (6(313,1X))
```

```
1030 FORMAT (20F4.1)
1040 FORMAT (10X4E15.8)
1050 FORMAT (10F8.6)
1060 FORMAT (10F8.1)
1070 FORMAT (10F8.3)
1080 FORMAT (10F8.5)
1090 FORMAT (0E12.6)
5000 FORMAT ( /, 2X, 12HFOR ELEMENT I3, 24H, THE BANDWIDTH HAS BEEN ,
             13HCHANGED FORM , 12, 4H TO , 12)
   1
5010 FURMAT (1H1, 2X, 9(14H-PLANE STRESS-))
5020 FORMAT ( //, 2X, 44HFOR REFERENCE, THE INPUT DECK IS REPRODUCED .
             15HIN ITS ENTIRETY , //, 4X, 16A5)
5030 FORMAT (1X, 2016)
5040 FORMAT (8(3x, 314))
5050 FORMAT (8(1x, 2F7,4))
5060 FORMAT (15X, 4E15.8)
5070 FORMAT (1X, 10F12.6)
5080 FORMAT (1X, 10F12.1)
5090 FORMAT (5X, 10F8.3)
6000 FORMAT (5X, 10F8.5)
0010 FURMAT ( 6(5%, 2814,6, / 1)
6020 FORMAT ( //2X9(14H-PLANE STRESS-))
     END
```

#### Subroutine Stifgn

```
SUBROUTINE STIFGN
  COMMON / CONSTI / NEJ, NSX, NSY, NANG, NFAIL, NXI, NX2, NYI, NY2
   COMMON / HATDAT / E11, E12, E13, E22, E23, E33, U(3,3), EE(3,3)
  COMMON / ANISOT / \varepsilon1, \varepsilon2, \varepsilon4, v12, v14 (5), ANILE(5), ALLOW(6,2), HT
  COMMON / COMPTI / ANGER(5), THETA(5), ITT(150,5)
   V21 = V12 * L2 / L1
   W(1,1) = E1 / (1,0 - V12 * V21)
  G(1,2) = V21 * G(1,1)
  w(1,5) = (82 *V12*V21) / (1.0 - V12*V21)
   u(2,1) = u(1,2)
   \omega(2,2) = E2 / (1.0 - V12 * V21)
  u(2,3) = (E1 *V12*V21) / (1.0 - V12*V21)
   u(3,1) = 0.0
   u(3,2) = 0.0
   u(3,3) = 2,0 * 6
   HT = 0.
  01 = .125 * (3. * (0(1.1) + 0(2.2)) + 2. * 0(1.2) + 2. * 0(3.3))
  U2 = .500 * (Q(1,1) - Q(2,2))
  03 = .125 * (0(1,1) + 0(2,2) - 2. * 0(1,2) - 2. * 0(3,3))
   04 = .125 * (0(1,1) + 0(2,2) + 0. * 0(1,2) - 2. * 0(3,3))
  U5 = .125 * (0(1,1) + 0(2,2) - 2. * (1,2) + 2. * (3,3))
   00 10 I = 1, NANG
   ANG2R(I) = ANGLE(I) * ((2.0 * 3.141592653) / 180.0)
10 HT = HT + THK(I)
   THE TRANSFORMED LAMINA STIFFNESS MATRIX (E) IS COMPUTED
  £11 = 0.
  E12 = 0.
  c13 = 0.
  E22 = 0.
```

```
£23 = 0.
     E33 = 0.
      DC 20 I = 1, NANG
     E11 = E11 + (U1 +U2*COS(ANG2R(I))+U3*COS(2,*ANG2R(I)))*THK(I)/HT
     E12 = E12 + (U4-U3+COS(2.*ANG2R(I)))*THK(I)/HT
     E13 = E13 + (0.50*U2*SIn(ANG2R(1))+U3*SIN(2,*ANG2R(1)))*THK(I)/HT
     E22 = E22 + (U1-U2*COS(AHG2R(I))+U3*COS(2.*ANG2R(I)))*THK(I)/HT
     E23 = E23 + (0.50+U2+SIN(ANG2R(1))-U3+SIN(2.+ANG2R(1)))+THK(1)/hT
      E33 = E35 + (U5-U3+CGS(2.*ANG2R(I)))*THK(I)/HT
  20 CONTINUE
     WRITE (6,5000)
     WHENTE (0,5010) E11, E12, E13, E22, E23, E33
     RETURN
5000 FORMAT (1H1, 15% 25HTHE COMPOSITE A-MATRIX IS)
5010 FORMAT (/15x3E15.5 // 30x2E15.5 // 45xE15.5)
     FIVE
```

#### Subroutine Constr

```
SUBROUTINE CONSTR
   CONSTR FIRST CALLS ELCONS WHICH CONSTRUCTS THE INDIVIDUAL ELEMENT
   STIFFACES MATRICES AND THEN USES THE ELEMENT MATRICES TO CONSTRUCT
   THE STIFFLESS MATRIX FOR THE WHOLE STRUCTURE.
   COMMON / ARRAYS / A(30,200), FO(200), FIN(200), DIS(200), TDIS(200)
   COMMON / GEOMET / IBD, NEL, HRD, NZC, NM(450), NZP(40), XYM(200)
   UO 10 J = 1,NRD
    FO(J) = 0,0
    00 10 1 = 1,180
    A(1.J) = 0.0
10 CONTINUE
   DO 20 I = 1.NEL
    13 = 1 + 3
    CALL ELCONS (13)
20 CONTINUE
   RETURN
   END
```

#### Subroutine Elcons

```
SUBROUTING ELCONS (13)
   COMMUN. / ARRAYS / A(30,200), FO(200), FIN(200), DIS(200), TDIS(200)
  COMMON / GEGMET / IBO, NEL, NRD, NZC, NM(45C), NZP(40), XYM(200)
  COMMON / MATUAT / E11, E12, E13, E22, E23, E33, G(3,3), EE(3,3)
   DIMENSION NOD (3)
   1400(1) = 140(13-2)
   1100(2) = 11M(13-1)
  MOU(3) = HM(13)
   00 20 1 = 1.2
    M = I + 1
    00 20 J = M,3
    IF (NOD(I) - NOD(J)) 20, 10, 10
    MT = NOU(1)
    NOD(I) = NOD(J)
    NOU(U) = NT
20 CONTINUE
```

```
BEST AVAILABLE COPY
```

```
N1Y = NOD(1) + 2
112Y = 1100(2) + 2
113Y = 100(3) * 2
N1X = N1Y - 1
N2X = N2Y - 1
113X = N3Y - 1
1121 = 112Y - N1Y
1,31 = N3Y - N1Y
N32 = N3Y - 112Y
X12 = XYM(N1Y-1) - XYM(N2Y-1)
\lambda 13 = XYM(N1Y-1) - XYM(N3Y-1)
X23 = XYM(N2Y-1) - XYM(N3Y-1)
11S = XYM(N1Y) - XYM(N2Y)
              ) - XYM(N3Y
Y13 = XYM(N1Y)
Y23 = XYM(N2Y) - XYM(N3Y)
AH4 = AB5(X12 * Y13 - X13 * Y12) * 2.0
SAR = SURT (AK4)
X12 = X12 / SAR
X13 = X13 / SAR
X23 = X23 / SAR
Y12 = Y12 / SAR
Y13 = Y13 / SAR
Y23 = Y23 / SAR
1723 = Y23 * Y23
xx23 = x23 * x23
XY23 = X23 * Y23
YY13 = Y13 * Y13
XX13 = X13 * X13
xY13 = X13 * Y13
YY12 = Y12 * Y12
XX12 = X12 * X12
xY12 = x12 * Y12
Y133 = Y13 * Y23
XX33 = X13 * X23
x33Y = x13 * Y23
Y33X = Y13 * X23
1122 = Y12 * Y23
XA22 = X12 * X23
x22Y = X12 * Y23
122X = 112 * 123
YY11 = Y12 * Y13
x \times 11 = x12 * x13
X11Y = X12 * Y13
Y11X = Y12 * X13
                     1.N1X) + E11*YY23 - 2.0*E13*XY23 + E33*XX23
AL
     1.N1X) = \Lambda(
                     2.111X) + E13+YY23 - (E12+E33)+XY23 +E25+XX23
A (
      2.1.1x) = A(
A(N21+1,N1X) = A(N21+1,N1X) + E15*X33Y-E11*YY35-E33*XX35*E15*Y53X
A(N21+2,N1X) = A(N21+2,N1X) + E12*X33Y-E13*YY33-E23*XX33+E35*Y33X
A(N31+1,N1X) = A(N31+1,N1X) + E11*YY22-E13*X22Y-L13*Y22X+E35*XX22
A(1,31+2,1,1x) = A(1,31+2,111x) + E13+YYZZ-E12+XZZY-E33+YZZX+EZ3+XXZZ
                    1,111Y) + E22+XX23 - 2.0+E23+XY23 + E33+YY23
     1,1,1Y) = A(
A(N21 (N1Y) = A(N21 (N1Y) + E12+Y33X-E23+XX33-E15+YY35+E33+X33Y
A(N21+1,N1Y) = A(N21+1,H1Y) + E25*Y33X-E22*XX35-E33*YY33+E23*X33Y
A(N31 /NIY) = A(N31 /NIY) + E23*XX22-E12+Y22X-E33*X22Y+E13+YY22
A(N31+1,N1Y) = A(N31+1,N1Y) + E22*XX22-E23*Y22X-E23*X22Y+E33*YY22
                   1.N2X) + E11*YY13 - 2.0*E13*XY13 + E33*XX13
AL
      1.N2X) = A(
                     2, NZX) + E13+YY13 - (E12+E33)+XY13+E23+XX13
AL
      2,N2X) = A1
A(N32+1,N2X) = A(N32+1,H2X) + E13*X11Y-E11*YY11-E33*XX11+E15*Y11X
A(N32+2,N2X) = A(N32+2,N2X) + E12*X11Y-E13*YY11-E23*XX11+E35*Y11X
     1/(2Y) = A( 1/(2Y) + E22*XX13 - 2.0*E23*XY13 + E33*YY13
A(1132 1127) = A(N32 1N2Y) + E12+Y11X-E23+XX11-E13+YY11+E33+X11Y
```

#### Subroutine Excite

```
SUBROUTINE EXCITE
    COMMON / ARRAYS / A(30,200), FO(200), FIN(200), DIS(200), TDIS(200)
    COMMON / GEOMET / IBD, NEL, NRD, NZC, NM(450), NZP(40), XYM(200)
    COMMON / XLOADS / NXD, NXF, NDPX(10), NFPX(10), NX(2,10), EX(2)
COMMON / YLOADS / NYD, NYF, NDPY(10), NFPY(10), NY(2,10), EY(2)
    131 = 180 - 1
    HOC = NXU + NYD
    IF (NDC .EQ. 0) GO TO 240 DO 130 1 = 1,NDC
     1F (1 - 1.XU) 10, 10, 20
 10 J = NUPX(I)
     U = EX(1)
     GO 10 30
     J = NOPY(I - NXU)
     D = EY(1)
 30
    IF (IBD - J) 40, 40, 60
     DO 50 K = 1, 181
      FO(J+K-IBD) = FU(J+K-IBD) - A(1-K+IBD, J+K-IBD) * D
    CONTINUE
     60 TO 80
    IF (U .E9, 1) 60 TO 60
UM = U - 1
      00 70 K = 1,JM
      FO(K) = FO(K) - \Lambda(J-K+1,K) * D
     CONTINUE
     IF (IBD - NRD + J) 90, 90, 110
      DO 100 K = 2.180
      FO(J+K-1) = FO(J+K-1) - A(K,J) * D
100
     CONTINUE
     60 10 130
     IF (J . LG. NRD) GO TO 130
     IL = 1410 - J + 1
      00 120 K = 2, IL
      FO(J+K-1) = FO(J+K-1) - A(K,J) * D
120 CONTINUE
130 CONTINUE
     DO 530 1 = 1'NDC
     IF (I - 1.XU) 140, 140, 150
    J = NDPX(I)
140
     D = EX(1)
     GO TO 160
150
     J = NOPY (I - NXU)
     0 = EY(1)
100
     IF (IGD - J) 170, 170, 190
170
      00 180 K = 1.181
      A(1-K+18D, J+K-18D) = 0.0
180
     CONTINUE
     GO TO 210
190
    IF (J .EG. 1) GO TO 210
```

```
JM = J - 1
      DU 200 K = 1,JM
      A(J-K+1.K) = 0.0
200
     CONTINUE
210
     00 220 K = 2.180
      A(K.J) = 0.0
   CONTINUE
    · FO(J) = A(1,J) * D
230 CONTINUE
240 NFC = NXF + NYF
    IF (NFC .EQ. 0) 60 TO 280
     00 270 1 = 1.NFC
1F (1 - NXF) 250, 250, 260
    J = NFPA(I)
     K = I_1 X(1,1) + 2
     L = NX(2, I) * 2
     0 = EX(2)
     5 = (XYM(L) - XYM(K)) / 2.0
     60 10 270
    J = NEPY (I - NXF)
     K = NY(1, 1 - NXF) * 2 - 1
     L = NY(2,1 - NXF) * 2 - 1
     0 = EY(2)
     S = (XYM(L) - XYM(K)) / 2.0
270 FU(J) = FO(J) + S * D
280 CONTINUE
     DO 340 1 = 1,NZC
     J = NZP(1)
     FO(J) = 0.0
     DO 290 K = 2, IBD
      A(K,J) = 0.0
290 CONTINUE
     1F (IBD - J) 300, 300, 320
     UO 310 K = 1, 181
      A(1-K+18D+J+K-180) = 0.0
310 CONTINUE
     60 TO 340
320
    IF (J .E9. 1) GO TO 340
     JM = J - 1
      DO 330 K = 1.JM
336
      A(1-K+J,K) = 0.0
346 CONTINUE
     DO 350 I = 1,NRD
     FIN(1) = FO(1)
350 CONTINUE
    RETURN
    END
```

#### Subroutine Gausel

```
SUBROUTINE GAUSEL
   COMMON / ARRAYS / A(30,200), FO(200), FIN(200), DIS(200), TDIS(200)
   COMMON / GEOMET / IBU, NEL, NRD, NZC, NM(450), NZP(40), XYM(200)
   182 = 180 - 2
   N1 = NRD - 182
   N2 = N1 + 1
    DC 20 N = 2.N1
    J2 = 182 + N
    DO 10 1 = N.J2
     R = A(1-N+2,N-1) / A(1,N-1)
     A(I-N+1,N) = A(I-N+1,N) - K * A(2,N-1)
    FO(1) = FO(1) - R * FO(11-1)
10 CONTINUE
    M = N + 1
     00 20 J = M, J2
      DU 20 1 = J.J2
      A(I-J+1,J) = A(I-J+1,J) - A(I-N+2,N-1)*A(J-N+2,N-1) / A(I,N-1)
20 CONTINUE
    DO 40 N = N2, NRD
     00 30 1 = N.J2
     R = A(1-N+2,N-1) / A(1,N-1)
     A(I-N+1,N) = A(I-N+1,N) - R * A(2,N-1)
     FO(1) = FO(1) - R * FO(N-1)
30 CONTINUE
    M = N + 1
    00 40 J = M.J2
      00 40 I = J.J2
      IF (J-NRD) 45, 45, 35
35
      A(I-J+1,J) = 0,0
45
      A(1-J+1,J) = A(I-J+1,J) - A(I-N+2,N-1)*A(J-N+2,N-1) / A(1,N-1)
40 CONTINUE
   00 60 J = 1.NRD
   R = 0.0
    1 = 1.RD - J
     00 50 K = 2, IBU
    IF (I+K-NRU) 55, 55, 80
80
   TDIS(I+K) = 0.0
55 R = K + A(K,I+1) + TDIS(I+K)
50 CONTINUE
    TDIS(I+1) = (FO(i+1) - R) / A(1,I+1)
60 CONTINUE
   RETURN
   END
```

#### Subroutine Elfail

```
SUBROUTINE ELFAIL
   COMMON / ARRAYS / A(30,200), FO(200), FIN(200), DIS(200), TDIS(200)
   COMMON / CONSTI / NES, HSX, HSY, NANG, NFAIL, HXI, NXZ, NYI, NY2
   COMMON / GEOMET / IBD, NEL, NRD, NZC, NM(450), NZP(40), XYM(200)
   COMMON / MATUAT / E11, E12, E13, E22, E23, E33, G(3,3), EE(3,3)
   COMMON / ANISOT / E1, E2, G, V12, THK(5), ANGLE(5), ALLOW(6,2), HT
   COMMON / COMPT1 / ANGER(5), THETA(5), ITT(150,5)
   COMMON / COMPT2 / ACC, NEM
   DIMENSION STRN(3), STRS(3), FOO(200)
    UU 4 1 = 1, NANG
    THETA(I) = ANGLE(I) * (3,14159 / 180.0)
4 CONTINUE
   INC = 1
 5 CONTINUE
    00 10 I = 1.NRD
    FO(1) = 0.0
10 CONTINUE
    00 30 I = 1.NEL
    13 = 1 + 3
    CALL STRAIN (13, EPX, EPY, GXY)
    CALL LAYER (13, EPX, EFY, GXY, LAY)
    IF (LAY .NE. 0) CALL PEORCE (13, EPX, EPY, GXY)
   IF (1 .NE, NEM) 60 TO 30 EE(1,1) = E11 - EE(1,1)
   EC(1,2) = E12 - EE(1,2)
   LE(1,3) = E13 - EE(1,3)
   EE(2,2) = E22 - EE(2,2)
   LE(2,3) = E23 - EE(2,3)
   LE(3,3) = E35 - EE(3,3)
   LE(2,1) = EE(1,2)
   EE(3,1) = EE(1,3)
   LE (3,2) = EE (2,3)
     UG 26 K = 1,3
     STKS(K) = 0.0
56 CONTINUE
    STRN(1) = EPX
    STRN(2) = EPY
    STRN(3) = GXY
     DO 27 K = 1,3
      CO 27 L = 1.3
      STAS(K) = STRS(K) + EE(K,L) * STRN(L)
27 CONTINUE
     DO 28 K = 1,3
      DO 20 L = 1,3
      EE(K,L)= 0.0
28 CONTINUE
30 CONTINUE
    LO 35 1 = 1.NZC
    J = 112P(1)
    FU(J) = 0.0
35 CONTINUE
  FOMAX = 0.0
    00 40 1 = 1.NRD
    F = FO(I)
    FO(I) = FO(I) - FOO(I)
    F00(I) = F
```

```
FOMAX = AMAX1(ADS(FU(I)), FOMAX)
 40 CONTINUE
   PRITE (0,45) INC. IIT, FOMAX
 45 FORMAT (15x, 11HINGREMNNT =, 13, 1H,, 2x, 12HITTERATION =, 13,
   11H., 2X, 7HEOMAX =, E11.4)
   IF (FCMAX .LT. ACC) 60 TO 50
    111 = 117 + 1
    IF (III .GT. 20) 60 TO 70
    CALL FOREDU
    60 TO 5
 50 CALL GUTPUT (INC. III)
    WRITE (6,75) NEM
    MRITE (6,80) (STRN(1), 1 = 1,3)
    WRITE (0,85) NEM
    WRITE (0,90) (STRS(1), I = 1,3)
    ARITE (6,100)
   M = NRD / 2
   00 55 I = 1,M
    J = 2 * 1 - 1
     K = 2 * I
     WRITE (6,110) I, TDIS(J), TDIS(K), FO(J), FO(K)
 55 CONTINUE
    WRITE (6,99)
    INC = INC + 1
     DO 60 I = 1,NRD
     FO(1) = FIN(1)
 60 CONTINUE
   CALL FOREDU
    III = 0
    60 TO 5
 70 CALL OUTPUT (INC. IIT)
   "RITE (6,75) NEM
    wRITE (0,80) (STHN(I), I = 1,3)
   WRITE (6,85) NEM
   WHITE (0,90) (STRS(I), I = 1,3)
   ARITE (6,100)
   M = NAD / 2
    CO 72 I = 1.M
    J = 2 * 1 - 1
    K = 2 * I
    WRITE (6,110) I, TDIS(J), TDIS(K), FO(J), FO(K)
 72 CONTINUE
75 FORMAT (1HO, 15x, 22HTHE STRAIN IN ELEMENT , 13, 2x, 3HIS:, //)
80 FORMAT (17X, 10HX-STRAIN =, E12.5, / 17X, 10HY-STRAIN =, E12.5,
  1/ 17X, 14HSHEAR STRAIN =, E12.5)
 65 FORMAT (1HQ, 15x, 22HTHE STRESS IN ELEMENT , 13, 2x, 3HIS:, //)
 90 FORMAT ($7X, 10HX-STRESS =, E12.5, / 17X, 10HY-STRESS =, E12.5,
  1/ 17x, 14h5HEAR STRESS =, E12.5)
100 FORMAT (1HO, 15x, 4HHODE, 4x, 6HX-DISP, 4x, 6HY-DISP, 5x,
  17HX-FORCE, BX, 7HI-FORCE, /)
 99 FORMAT (1H1)
110 FORMAT (16x, I3, 2(4x, F6,4), 2(4x, E11,4))
   KETURN
   END
```

#### Subroutine Strain

```
SUBROUTINE STRAIN (13, EPX, EPY, GXY)
COMMON: / ARRAYS / A(30,200), FO(200), FIN(200), DIS(200), TDIS(200)
COMMON / GEOMET / IBD', NEL, NRD, NZC, NM(450), NZP(40), XYM(200)
111Y = NM(13-2) + 2
N2Y = NM(13-1) * 2
113Y = 11M(13) + 2
X12 = XYM(N1Y-1) - XYM(N2Y-1)
x13 = XYM(N1Y-1) - XYM(N3Y-1)
x = x \times (1 - x \times 1) = x \times (1 - x \times 1)
Y12 = XYM(N1Y) - XYM(N2Y)
Y13 = XYM(N1Y) - XYM(N3Y)
YEN MYX - ( YSN) MYX = 684
AR2 = ABS(X12*Y13 - X13*Y12)
U1 =TUIS(N1Y-1)
U2 =TDIS(N1Y )
03 = (DIS (N2Y-1)
U4 = 1015 (112Y )
D5 = TDIS (N3Y-1)
U6 =TUIS(113Y
EPX = (Y23*D1 - Y13*D3 + Y12*D5) / AR2
EPY =- (X23*D2 - X13*D4 + X12*D6) / AR2
GXY = (Y23+D2 - Y13+D4 + Y12+D6 - X23+D1 + X13+D3 - X12+D5) / AR2
RETURN
END
```

#### Subroutine Layer

```
SUBROUTINE LAYER (13, EPX, EPY, GXY, LAY)
   COMMON / CONSTI / NE3, MSX, MSY, MANG, MEAIL, MXI, MXZ, MYI, MYZ
   COMMON / MATUAT / E11, E12, E13, E22, E23, E33, Q(3,3), EE(3,3)
   COMMON / ANISOT / E1, E2, G, V12, THK(5), ANGLE(5), ALLCH(6,2), HT
   COMMON / COMPT1 / ANG2R(5), THETA(5), ITT(150,5)
   REAL ESTRI(3), LASTRI(3), LASTRS(3), T(3,3)
   ESTRIN(1) = EPX
   ESTRIV(2) = EPY
   ESTRN(3) = GXY / 2.0
   LAY = 0
   J'= 13 / 3
    UO 151 K = 1. NANG
    1TT(J,K) = 1
    T(1,1) = (COS(THETA(K))) * (COS(THETA(K)))
    I(1,2) = (SIN(THETA(K))) * (SIN(THETA(K)))
    T(1,3) = 2.0 * SIN(THETA(K)) * COS(THETA(K))
    T(2,1) = T(1,2)
    T(2,2) = T(1,1)
    I(2,3) = -I(1,3)
    T(3,1) = - SIN(THETA(K)) * COS(THETA(K))
    T(3,2) = -1(3,1)
    T(3,3) = T(1,1) - T(1,2)
     00 10 1 = 1.3
     LMSTRN(1) = 0.0
      00 10 L = 1.3
      LMSTRN(I) = LMSTRN(I) + T(I \cdot L) * ESTRN(L)
10 CONTINUE
```

```
00 20 1 = 1.2
      LMSTRS(1) = 0.0
       DO 20 L = 1,2
       LMSTRS(I) = LMSTRS(I) + Q(I,L) * LMSTRN(L)
    CONTINUE
     LMSTRS(3) = G(3,3) * LMSTRN(3)
      00 141 1 = 1.3
      IF (NFAIL .EQ. 2) 60 TO 50
      IF (LMSTRN(I) .LT, 0.0) GO TO 40
      IF (LMSTRN(I) - ALLOW(1.1)) 140, 140, 70
      IF (LMSTRN(1) + ALLOW(1,2)) 70, 140, 140
 40
 50
      CONTINUE
      L = I + 3
      IF (LMSTRS(I) .LT, 0.0) GO TO 60
IF (LMSTRS(I) - ALLOW(L,1)) 140, 140, 70
IF (LMSTRS(I) + ALLOW(L,2)) 70, 140, 140
 00
 70
      CONTINUE
      GO TO (80, 90, 100), I
 80
      ITI(J,K) = 2
      GO TO 140
 90
      IF (ITT(J.K) .Eu. 1) GU TO 95
      ITI(J,K) = 5
      GO TO 140
      ITI(J,K) = 3
      GO TO 140
100
      M = ITT(J,K)
      GO TO (105, 100, 107, 140, 108), M
105
      ITT(J,K) = 4
      GO 10 140
106
      1111Jih) = 6
      GO TO 140
107
      ITT(J,K) = 7
      GO TO 140
108
      ITT(J,K) = 8
140
     CONTINUE
141
     CONTINUE
     IF (ITT(J.K) .EG. 1)GO TO 150
     N = ITT(U,K)
     LAY = LAY + 1
     IF (LAY .GT. 1) GO TO 145
      00 142 1 = 1,3
       00 142 1. = 1,5
       EE(I,L) = 0.0
142 CONTINUE
145 CONTINUE
     CALL DESTIF (N.K)
150 CONTINUE
151 CONTINUE
    EE(2,1) = EE(1,2)
    EE(3,1) = EE(1,3)
    EE(3,2) = EE(2,3)
    KETUKN
    END
```

#### Subroutine D1stif

```
SUBROUTINE DESTIF (N.K)
   CUMMON / MATUAL / E11, E12, E13, E22, E23, E33, G(3,3), EE(3,3)
    COMMON / ANISOT / E1, E2, G, V12, THK(5), ANGLE(5), ALLO, (6,2), HT
    COMMON / COMPTI / ANGZR(5), THETA(5), ITT(150,5)
    GO TO (100, 10, 20, 20, 30, 30, 20, 30), N
 10 01 = 125 * (3. * (0(1:1) + 0(1:5)) + 2. * 0(1:2) + 2. * 0(3:3))
    U2 = .500 + (G(1,1) - G(1,3))
    U3 = .125 * (Q(1,1) + Q(1,3) - 2. * Q(1,2) - 2. * Q(3,3))
    04 = .125 * (9(1,1) + 9(1,3) + 6. * 9(1,2) - 2. * 9(3,3))
    05 = .125 * (0(1,1) + 0(1,3) - 2. * 0(1,2) + 2. * 0(3,3))
   GO TO 40
 20 U1 = .125 * (3, + (G(2.3) + G(2.2)) + 2, * G(1.2) + 2, * G(3.3))
    U2 = .500 * (9(2,3) - 9(2,2))
    03 = .125 * (0(2.3) + 0(2.2) - 2. * 0(1.2) - 2. * 0(3.3))
    04 = .125 * (0(2,3) + 0(2,2) + 6. * 0(1,2) - 2. * 0(3,3))
    05 = .125 * (0(2,3) + 0(2,2) - 2, * 0(1,2) + 2, * 0(3,3))
    60 TO 40
 30 \text{ U1} = .125 * (3. * (9(1.1) + 9(2.2)) + 2. * 9(1.2) + 2. * 9(3.3))
    UZ = .500 * (Q(1,1) - Q(2,2))
    03 = .125 * (0(1,1) + 0(2,2) - 2. * 0(1,2) - 2. * 0(3,3))
    04 = .125 * (0(1.1) + 0(2.2) + 0. * 0(1.2) - 2. * 0(3.3))
    05 = .125 * (0(1.1) + 0(2.2) - 2. * 0(1.2) + 2. * 0(3.3))
 40 CULTINUE
    EE(1,1)=EE(1,1)+(U1+U2*CUS(ANG2R(K))+U3*COS(2,*ANG2R(K)))*THK(K)
   EE(1,2)=EE(1,2)+(U4-U3+CUS(2,*ANG2R(K)))*THK(K)/HT
   EE(1,3)=EE(1,3)+(0,50+U2*SIN(ANG2R(K))+U3*SIN(2,*ANG2R(K)))*THK(K)
   1/HT
   EE(2,2)=EE(2,2)+(U1-U2+CUS(ANG2R(K))+U3+COS(2.*ANG2R(K)))*THK(K)
   1/HT
   EE(2,3)=EE(2,3)+(0.50*U2*SIN(ANG2R(K))-U3*SIN(2.*ANG2R(K)))*THK(K)
   EE(3,3)=EE(3,3)+(U5-U3*CUS(2,*ANG2R(K)))*THK(K)/HT
100 CONTINUE
   KETURN
   END
```

AD-A043 749

AIR FORCE INST OF TECH WRIGHT-PATTERSON AFB OHIO F/G 11/4
PROGRESSIVE FAILURE OF ADVANCED COMPOSITE LAMINATES USING THE F--ETC(U)
MAR 76 G E BROWN
AFIT-CI-77-59 NL

UNCLASSIFIED

AD A043 749











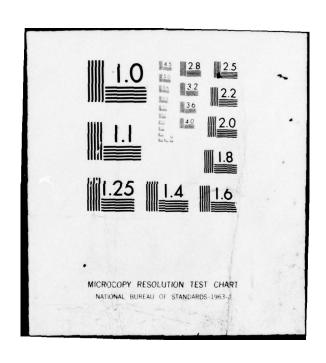






END DATE FILMED

9-77 DDC



#### Subroutine Pforce

```
SUBROUTINE PHORCE (13, EPX, EPY, GXY)
   COMMON / ARRAYS / A(30,200), FO(200), FIN(200), DIS(200), TDIS(200)
   COMMON / GEORET / IBD. HEL. NRD. NZC. NM(450), NZP(40), XYM(200)
   COMMON / MATURI / ELL. ELZ, ELZ, EZZ, EZZ, EZZ, G(3,3), EE(3,3)
   COMMON / ANISOT / E1, E2, G, V12, THK(5), ANGLE(5), ALLOW(6,2), HT
   DIMENSION B(6,3), 68(6,3), EP(3)
    00 10 1 = 1.6
     UO 10 J = 1,3
     BB(I.J) = 0.0
     B (I,J) = 0,0
16 CONTINUE
   m1Y = NM(13-2) + 2
   N2Y = NM(13-1) * 2
   13Y = NM(13 ) * 2
   3(5,3) = XYM(N2Y-1) - XYM(N1Y-1)
   6(3,3) = XYM(N1Y-1) - XYM(N3Y-1)
   o(1.5) = XYM(N3Y-1) - XYM(N2Y-1)
   \begin{array}{lll} E(5,1) &= XYM(H1Y) &= XYM(H2Y) \\ E(5,1) &= XYM(H1Y) &= XYM(H2Y) \\ E(5,1) &= XYM(H3Y) &= XYM(H1Y) \end{array}
   b(1,1) = XYM(42Y ) - XYM(N3Y )
   8(2,3) = 8(1,1)
   b(2,2) = B(1,3)
   B(4,3) = b(3,1)
   b(4,2) = B(3,3)
   B(6,3) = B(5,1)
   L(6,2) = E(5,3)
    00 20 I = 1,6
     00 20 K = 1,3
      DO 20 L = 1.3
      68(I.K) = 66(I.K) + 8(I.L) * EE(L.K)
20 CONTINUE
   LP(1) = EPX + HT / 2.0
   LP(2) = LPY * HT / 2.0
   EP(3) = GXY * HT / 2.0
    00 30 1 = 1.3
    FO(HIY-I) = FO(HIY-I) + BB(I,I) * EP(I)
    FO(N1Y) = FO(N1Y) + UB(2.1) * EP(1)
    FO(N2Y-1) = FO(N2Y-1) + EB(3,1) * EP(1)
    FO(N2Y ) = FO(N2Y ) + BB(4,1) * EP(1)
    FO(N3Y-1) = FO(N3Y-1) + BB(5,1) + EP(1)
    FO(113Y ) = FO(NSY ) + 68(0,1) * EP(1)
30 CONTINUL
   KETUKN
   END
```

#### Subroutine Foredu

```
SUBROUTINE FOREDU
   COMMON / ARRAYS / A(30,200), FO(200), FIN(200), DIS(200), TDIS(200)
COMMON / GEOMET / IBD, NEL, NRD, NZC, NM(450), NZP(40), XYM(200)
   182 = 180 - 2
   N1 = NRU - IB2
   N2 = N1 + 1
    DO 10 N = 2,N1
    J2 = IU2 + N
     00 10 1 = N.J2
     FO(1) = FO(1) - (A(1-N+2,N-1) / A(1,N-1)) * FO(N-1)
10 CONTINUE
    00 20 N = N2, NRD
     DO 20 1 = N.J2
     FO(I) = FO(I) - (A(I-N+2,N-1) / A(1,N-1)) * FO(N-1)
20 CONTINUE
    00 60 J = 1,NRD
    R = 0.0
    I = NRU - J
     60 50 K = 2, IBU
     IF (I+K-NRU) 40, 40, 30
30
    DIS(I+K) = 0.0
     R = R + A(K,I+1) * DIS(I+K)
50 CONTINUE
    UIS(I+1) = (FO(I+1) - R) / A(1,I+1)
60 CONTINUE
    00 70 I = 1.NRU
    TDIS(I) = TDIS(I) + DIS(I)
70 CONTINUE
   RETURN
   END
```

#### Subroutine Output

```
SUBROUTINE OUTPUT (INC. IIT)
    CUMMUN / CONSTI / NE3, NSX, NSY, NANG, NFAIL, NXI, NX2, NY1, NY2
    COMMON / GEOMET / IBU, NEL, NRD, NZC, NM(450), NZP(40), XYM(200)
    COMMON / ALOADS / NXU, HAF, LEPX(10), NFPX(10), HX(2,10), EX(2)
    COMMON / YLOAUS / NYU. NYF, NOPY (10), NFPY (10), 1.Y(2,10), EY (2)
    COMMON / ANISOT / E1, 22, G, V12, THK(5), ANGLE(5), ALLOW(6,2), HT
    CUMMON / COMPTI / ANGER(5), THETA(5), ITT(150,5)
    IF (IIT .6T. 20) WRITE (6,100) IIT, INC
IF (IIT .LE. 20) WRITE (0,170) INC, IIT
IF (NSX .EG. 1) GO TO 10
    TALOAD = EX(1) * FLOAT(INC)
    TYLOAU = EY(1) * FLOAT(INC)
    WHITE (6,110) INC, TXLOAD, TYLOAD
    60 TO 20
 10 TXLOAD = EX(2) * FLOAT(INC)
    TYLOAD = EY(2) * FLOAT(INC)
    WRITE (6,120) INC, TXLOAD, TYLOAD
 20 "KITE (0,130)
    WRITE (6,140)
    WRITE (0,150) (ANGLE(I), I = 1,5)
     00 30 I = 1.NEL
     WRITE (0,160) I, (ITT(I,J), J = 1, NANG)
 30 CONTINUE
100 FORMAT (1HO, 15x, 29HTHE LAMINATE HAS FAILED AFTER, 13, / 15x,
             17HITERATIONS IN THE, 14, 1X, 10HINCREMENT.)
110 FORMAT (1H , 15x, 23HTHE MAGNITUDE OF THE APPLIED / 15x,
             E4HDISPLACEMENT THROUGH THE: 14: / 15X; 13HINGREMENT WAS;
1 F6.4, OHINCHES, / 15x, 22HIN THE X DIRECTION AND, F6.4,
1 / 15x, 20HINCHES IN THE Y DIRECTION.)
120 FORMAT (1H, 15x, 28HTHE MAGNITUDE OF THE APPLIED / 15x,
             16HLOAD THROUGH THE, 14, 1X, 9HINCREMENT / 15X, 3HWAS,
             F11.0, 1x, 14HLb/IN ON THE X / 15X, 8HFACE AND, F11.0, 1X,
             12HLB/IN ON THE / 15X, 7HY FACE.)
130 FORMAT (1HO, 15x, 13HFAILURE CCDE: / 17x, 14H1 = NO FAILURE / 17x,
            29H2 = FAILED PARALLEL TO FIBERS / 17x, 34H3 = FAILED PERPI
   INDICULAR TO FIBERS / 17X, 19H4 = FAILED IN SHEAR / 17X, 47H5 = FAI
   1LED PARALLEL AND PERPINDICULAR TO FIBERS / 17X, 42H6 = FAILED PARA
   ILLEL TO FIBERS AND IN SHEAR / 17x, 47H7 = FAILED PERPINDICULAR TO
   IFIBERS AND IN SHEAR / 17x, 29H8 = FAILED IN ALL THREE MODES / 17x,
   1 29H5.6.8 REPRESENT TOTAL FAILURE //)
146 FORMAT (22X, 48HLAMINA 1 LAMINA 2 LAMINA 3 LAMINA 4 LAMINA 5)
150 FORMAT (15x, THELEM , 5(4HANG= , F4.0, 2x), /)
160 FORMAT (15X, 13, 7X, 5( 11, 9X))
170 FORMAT (15X, 11HTHIS IS THE, 14, 1X, 10HINCREMENT. / 15X,
             27HEGUILIBRIUM WAS OBTAINED IN, 13, 1X, 11HITERATIONS, /)
    RETURN
    ENU
```

#### LIST OF REFERENCES

- <sup>1</sup>Jones, Robert M., <u>Mechanics of Composite Materials</u>, Mc-Graw-Hill Book Co., New York, N. Y., 1975, pp. 1-239.
- <sup>2</sup>Garg, Sabodh K., Svalbonas, Uytas, and Gartman, Gerald A., Analysis of Structural Composite Materials, Marcel Dekker, Inc., New York, N. Y., 1973, pp. 1-9, 71-80, 201-271, 287-319.
- Ashton, J. E., Halpin, J. C., Petit, P. H., <u>Primer on Composite Materials: Analysis</u>, Technomic Publishing Co., Inc., Stamford, Connecticut, 1969, pp. 1-58.
- "Schwartz, H. S., "Applications of Reinforced Plastics in Aircraft," <u>Proceedings of the Fifth Symposium on Naval Research</u>, Office of Naval Research, 1967, pp. 113-127.
- McCullouth, R. L., <u>Concepts of Fiber-Resin Composites</u>, Marcel Dekker, Inc., New York, N. Y., 1971, pp. 9-15.
- <sup>6</sup>Ashton, J. E., Burdorf, M. L., and Olson, F., "Design Analysis and Testing of an Advanced Composite F-111 Fuselage," <u>Composite</u> Materials: <u>Testing and Design</u>. (Second Conference), ASTM STP 497, Am. Soc. Testing Materials, 1972, pp. 3-27.
- <sup>7</sup>Heldenfels, R., "Applications of Composite Materials in Space Vehicle Structures," <u>Proceedings of the Fifth Symposium on Naval Structural Mechanics</u>, Pergamon, New York, 1967, pp. 157-174.
- <sup>8</sup>McClintock, Frank A., and Argon, Ali S., <u>Mechanical Behavior of Materials</u>, Addison-Wesley Publishing Company, Inc., Reading, <u>Massachusetts</u>, 1966, pp. 96-141, 488-490.
- <sup>9</sup>Chamis, C. C., "Micromechanics Strength Theories," <u>Composite</u> <u>Materials</u>, Vol. 5, Academic Press, New York, N. Y., 1974, pp. 93-148.
- <sup>1</sup> Chou, T. S., and Hermans, J. J., "The Elastic Constants of Fiber Reinforced Materials," Journal of Composite Materials, Vol. 3, 1969, p. 382.
- <sup>1</sup>Rosen, B. W., "A Simple Procedure for Experimental Determination of the Longitudinal Shear Modulus of Unidirectional Composites," Journal of Composite Materials, Vol. 6, 1972, p. 552.

- <sup>12</sup>Griffith, J. E., and Baldwin, W. M., "Failure Theories of Generally Orthotropic Materials," <u>Developments in Theoretical and Applied Mechanics</u>, Vol. 1, Plenum Press, New York, N. Y., 1963, pp. 410-420.
- <sup>13</sup>Hoffman, O., "The Brittle Strength of Orthotropic Materials," Journal of Composite Materials, Vol. 1, 1967, pp. 200-206.
- <sup>14</sup>Tsai, Stephen W., and Azzi, Victor D., "Strength of Laminated Composite Materials," AIAA Journal, Vol. 4, Feb. 1966, pp. 296-301.
- <sup>15</sup>Haisler, Walter E., and Stricklin, James A., "Development and Evaluation of Solution Procedures for Geometrically Nonlinear Structural Analysis," AIAA Journal, Vol. 10, No. 3, Mar. 1972, pp. 264-272.
- <sup>16</sup>Mallett, R. H., and Marcal, P. V., "Finite Element Analysis of Nonlinear Structure," Journal of the Structural Division, Vol. 94, No. ST9, Sept. 1968, pp. 2081-2105.
- <sup>17</sup>Stricklin, J. A., Haisler, W. E., and Von Riesemann, W. A., "Geometrically Nonlinear Analysis by the Direct Stiffness Method," Journal of the Structural Division, Vol. 97, No. ST9, Sept. 1971.
- <sup>18</sup>Nayak, G. C., and Zienkiewicz, O. C., "Elasto-Plastic Stress Analysis: A Generalization for Various Constitutive Relations Including Strain Softening," *International Journal for Numerical Methods in Engineering*, Vol. 5, 1972, pp. 113-135.
- <sup>19</sup>Zienkiewicz, O. C., Valliappan, S., and King, I. P., "Elasto-Plastic Solutions of Engineering Problems: Initial Stress Finite Element Approach," International Journal for Numerical Methods in Engineering, Vol. 1, 1969, pp. 75-100.
- <sup>20</sup>Hofmeister, L. D., Greenbaum, G. A., and Evensen, D. A., "Large Strain Elasto-Plastic Finite Element Analysis," AIAA Journal, Vol. 9, No. 7, July 1971, pp. 1248-1254.
- <sup>21</sup>Zienkiewicz, O. C., Valliapan, S., and King, I. P., "Stress Analysis of Rock as a 'No Tension' Material," Geotechnique, Vol. 18, 1968, pp. 56-66.
- <sup>22</sup>Zienkiewicz, O. C., and Cheung, Y. K., <u>The Finite Element Method in Structural and Continuum Mechanics</u>, McGraw-Hill Publishing Co., London, England, 1967, pp. 26-36, 192-211.
- <sup>23</sup>Zienkiewicz, O. C., <u>The Finite Element Method in Engineering Science</u>, McGraw-Hill Publishing Co., London, England, 1971, pp. 48-59, 369-392.
- <sup>2</sup> Cook, Robert D., <u>Concepts and Applications of Finite Element Analysis</u>, John Wiley and Sons, Inc., New York, N. Y., 1974, pp. 1-50, 293-308.

- <sup>25</sup>Zienkiewicz, O. C., and Irons, B. M., "Matrix Iteration and Acceleration Processes in Finite Element Problems of Structural Mechanics," <u>Numerical Methods For Nonlinear Algebraic Equations</u>, Gordon and Breach Science Publishers, New York, N. Y., 1970, pp. 183-194.
- <sup>26</sup>Tillerson, J. R., Stricklin, J. A., and Haisler, W. E., "Numerical Methods for the Solution of Nonlinear Problems in Structural Analysis," <u>Numerical Solution of Nonlinear Structural Problems</u>, ASME, AMD-Vol. 6, 1973, pp. 67-101.
- <sup>27</sup>Gallagher, Richard H., <u>Finite Element Analysis</u>, Prentice-Hall, Inc., Englewood Cliffs, New Jersey, 1975, pp. 118-125.
- <sup>28</sup>Organick, Elliot I., <u>Fortran IV</u>, Addison-Wesley Publishing Co., Reading, Mass., 1966, pp. 3-293.
- <sup>29</sup>Sendeckyj, G. P., Richardson, M. D., Pappas, J. E., "Fracture Behavior of Thoronel 300/5208 Graphite-Epoxy Laminates, Part I; Unnotched Laminates," AFFDI-TM-74-89-FBC, Structures Division, Air Force Flight Dynamics Laboratory, Wright-Patterson A.F.B., Ohio, 1974, pp. 3-21.
- <sup>30</sup>Nuismer, R. J., and Whitney, J. M., "Uniaxial Failure of Composite Laminates Containing Stress Concentrations," ASTM, STP 593, 1975.
- <sup>31</sup>Zweben, Carl, "Fracture Mechanics and Composite Materials: A Critical Analysis," ASTM, STP 521, American Society for Testing Materials, 1973, pp. 65-97.
- <sup>32</sup>Zweben, Carl., "An Approximate Method of Analysis for Notched Unidirectional Composites," Engineering Fracture Mechanics, Vol. 6, 1974, pp. 1-10.
- <sup>33</sup>Konish, A. J., Jr., Swedlow, J. L., and Crase, T. A., "Experimental Investigation of Fracture in an Advanced Fiber Composite," Journal of Composite Mar. <sup>1-1</sup>s, Vol. 6, 1972, pp. 114-124.
- <sup>3</sup> Konish, J. J., J. Cruse, T. A., "The Determination of Fracture Strength in Oruno copic Graphite-Epoxy Laminates," Air Force Contract F33615-73-C-5505, Air Force Materials Laboratory, Wright-Patterson, A.F.B., Ohio.
- <sup>35</sup>Waddoups, M. E., Eisenmann, J. R., and Kaminski, B. E., "Macroscopic Fracture Mechanics of Advanced Composite Materials," Journal of Composite Materials, Vol. 5, 1971, pp. 446-454.
- <sup>36</sup>Sandhu, R. S., "Ultimate Strength Analysis of Symmetric Laminates," AFFDL-TR-73-137, Wright-Patterson A.F.B., Ohio.
- <sup>3</sup> <sup>7</sup>Pagano, N. J., and Pipes, R. B., Journal of Composite Materials, Vol. 6, 1972, pp. 114-124.

#### VITA

NAME:

Gary Earl Brown

BIRTHPLACE:

Kingston, Utah

BIRTHDATE

11 November 1941

HIGH SCHOOL:

South Sevier High School

Monroe, Utah

COLLEGE:

1960-1961, 1963-1964 College of Southern Utah

Cedar City, Utah

UNIVERSITY:

1964-1967

Brigham Young University

Provo, Utah

1974-1976

University of Utah Salt Lake City, Utah

DEGREES:

1964

Associate of Science College of Southern Utah

Cedar City, Utah

1967

B.E.S., Mechanical Engineering

Brigham Young University

Provo, Utah

1976

M.S., Mechanical Engineering

University of Utah Salt Lake City, Utah

PROFESSIONAL ORGANIZATIONS:

Tau Beta Pi, American Society of

Mechanical Engineers

PROFESSIONAL POSITIONS:

1969-1973, Instructor Pilot 3525th Pilot Training Squadron

Williams A.F.B., Arizona

1973-1974, Wing Life Support Officer 374th Tactical Airlift Wing, Clark A.F.B.

Republic of the Philippines