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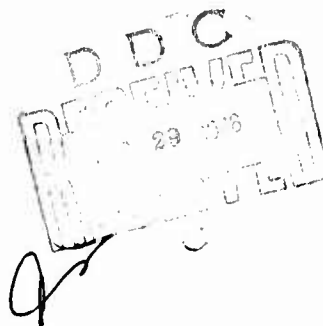
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CONTRACT REPORT NO. 295

FRACTURE MODELING IN THE SMITE CODE

Prepared by

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March 1976

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REPORT DOCUMENTATION PAGE		READ INSTRUCTIONS BEFORE COMPLETING FORM	
1. REPORT NUMBER BRL CONTRACTOR REPORT NO. 295	2. GOVT ACCESSION NO.	3. RECIPIENT'S CATALOG NUMBER	
4. TITLE (and Subtitle) Fracture Modeling in the SMITE Code.	5. TYPE OF REPORT & PERIOD COVERED Final rept.		
7. AUTHOR(s) S. Z. Burstein H. S. Schechter	6. PERFORMING ORG. REPORT NUMBER P-7111		
9. PERFORMING ORGANIZATION NAME AND ADDRESS Mathematical Applications Group, Inc. 3 Westchester Plaza Elmsford, NY 10523	8. CONTRACT OR GRANT NUMBER(s) DAAD05-75-C-0741 NEW		
11. CONTROLLING OFFICE NAME AND ADDRESS US Army Ballistic Research Laboratories Aberdeen Proving Ground, MD 21005	10. PROGRAM ELEMENT, PROJECT, TASK AREA & WORK UNIT NUMBERS		
14. MONITORING AGENCY NAME & ADDRESS (if different from Controlling Office) US Army Materiel Development and Readiness Command 5001 Eisenhower Avenue Alexandria, VA 22333	12. REPORT DATE MARCH 1976		
	13. NUMBER OF PAGES 28		
	15. SECURITY CLASS. (of this report) UNCLASSIFIED		
16. DISTRIBUTION STATEMENT (of this Report) Approved for public release; distribution unlimited.			
17. DISTRIBUTION STATEMENT (of the abstract entered in Block 20, if different from Report)			
18. SUPPLEMENTARY NOTES			
19. KEY WORDS (Continue on reverse side if necessary and identify by block number) nucleation crack growth brittle fracture ductile failure numerical modeling			
20. ABSTRACT (Continue on reverse side if necessary and identify by block number) This report documents the incorporation of the Stanford Research Institute's NAG-FRAG model into the SMITE code, a second order, Eulerian finite difference scheme modeling elasto-plastic material response to impact. Sample computations are presented.			

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I. INTRODUCTION

The reader is assumed to be familiar with the axisymmetric version of the SMITE code which is described fully in BRL CR-255. It is also assumed that the reader is familiar with the Stanford Research Institute (SRI) ductile fracture subroutine, DFRAC^T, and brittle fracture subroutine, BFRAC^T. The description of the theory of fracture used in these routines are contained in AMMRC CTR 75-2 and BRL CR-222.

The purpose of this research effort was twofold:

(i) First to modify the SMITE code so that the ductile and brittle fracture model (that of SRI) could be incorporated into SMITE.

(ii) Compare results obtained with the modified code with the Lagrangian formulation of SRI.

The modifications introduced into the SMITE code to incorporate the fracture subroutines are described in the following section. Each subroutine that required changes is described.

II. FRACTURE MODIFICATIONS TO SMITE

DENSB - Since pressure is now carried as an explicit variable, the pressure on the free surface is set to that necessary to produce zero normal stress. The iteration of density to obtain the required pressure through the solid equation of state is removed.

FDIFF - Logic is added to call IFRACT and hence obtain the relative properties from the fracture subroutines once the critical pressure has been exceeded at a point.

IFRACT - This is the routine that actually incorporates the fracture subroutine DFRACT or BFRACT. This is accomplished by introducing a quasi-Lagrangian calculation into the Eulerian code. Once the velocities at the advanced time are obtained at a mesh point by the two step method, they are integrated backwards in time to determine where that mesh point came from in a Lagrangian sense. Values of all variables at the initial time are then assigned to this point by linear interpolation from surrounding points. The ductile fracture subroutine DFRACT or the brittle fracture subroutine BFRACT may then be called to update these values to the original mesh point at the advanced time.

Values for strain are obtained by numerically differentiating the velocities using central differences. An α acceleration at a point with index i , for instance, may be obtained by differencing velocities at points $i-1$ and $i+1$. If the Lagrangian point lies between mesh lines i and $i+1$, acceleration at $i+1/2$ may also be obtained by differencing velocities at points i and $i+1$. Similar differencing is used to obtain β accelerations. Accelerations at the Lagrangian point are then evaluated by interpolation. The necessary strains are obtained by transforming from α, β to z, r space.

INFACE - The pressure on both sides of the interface is set to that necessary to produce a continuous normal stress. The iteration of density to obtain the required pressure through the solid equation of state is removed.

INITAL - The number of dependent variables is changed to 10 for ductile fracture and 22 for brittle fracture. Seven parameters are read in for ductile fracture and fourteen for brittle fracture.

INTRPL - Coding is added to reflect the additional variables about the axis of symmetry.

OUTPUT - Coding is added to print the additional variables at points where fracture has occurred.

PLTOUT - The output of points that are in the plastic region is removed and the amount of damage to fractured points is substituted.

PRS - The solid equation of state is changed to the Mie-Grüneisen equation.

$$P = (C\mu + D\mu^2 + S\mu^3) \left(1 - \frac{\Gamma\mu}{2}\right) + \rho p E$$

where C, D, S are constants, Γ is the Gruneisen ratio, E is the internal energy, ρ is the density and $\mu = \rho/\rho_0 - 1$. In the next section, we describe the modifications required for input data preparation.

III. INPUT

There are three types of input data, integer (I), real (R) and alphanumeric (A). An integer is a number without a decimal point which must be right justified in its field. A real number has a decimal point which may be followed by an exponent of the form $E+N$. The $+N$ represents the power of 10 by which the number is to be multiplied. The + may be omitted if N is positive and the E need not appear if either the + or - is present. If an exponent is present it must be right justified in the field. Alphanumeric input consists of exactly the punched characters and blanks appearing in the field. A description of the necessary input data for SMITE follows. The card number refers to the type of input. If more than one card is necessary for this input, there may be several cards of the same type number.

CARD	COLS.	TYPE	NAME	DESCRIPTION
1	1-80	A	TITLE	Any alphanumeric information to be printed at the beginning of the output.
2	1-10	R	TMAX	Maximum problem time in microseconds.
	11-20	R	TPRIN	Print output increment in microseconds.
	21-30	R	TPRPL	Printer plot output increment in microseconds.
	31-40	R	TPLOT	Plot tape output increment in microseconds.
	41-50	R	TSAVE	Restart tape output increment in microseconds. If any output increment is negative that output is suppressed.
	51-60	R	TCOMP	Maximum computation time in seconds.
	61-70	R	PORG	Value in centimeters of axial coordinate at left center of printer plot. The radial coordinate is set to zero.
	71-80	R	PSCL	Scale of printer plot in centimeters per inch.
3	1-10	R	RDIS	Minimum fractional mesh allowed. If a point is closer to the boundary than this fraction of mesh size, interpolation will be used.
4	1-5	I	NMAT	Number of materials.
	6-10	I	ISTART	Restart flag. Zero for initial run, otherwise non zero.
5	1-5	I	NINFC	Number of material interfaces. Use 0 if there is only one material and omit card 6.
6	1-5	I	INFC(1)	Index of dominant material of interface. Boundary of this material defines interface.
	6-10	I	INFC(2)	Index of first boundary point on interface.
	11-15	I	INFC(3)	Index of last boundary point on interface.
	16-20	I	INFC(4)	Index of second material of interface. Interface boundary of this material replaced by interface boundary of first material.
	21-25	I	INFC(5)	Index of last boundary point on interface.
	26-30	I	INFC(6)	Index of first boundary point on interface.
	31-35	I	INFC(7)	Index of dominant material of next interface, etc.
	36-40	I	INFC(8)	
	41-45	I	INFC(9)	This sequence is repeated, six entries per interface, two interfaces per card until all interfaces have been defined. The boundary point indices refer to the points defining the material boundaries in the order in which they are input.
	46-50	I	INFC(10)	
	51-55	I	INFC(11)	
	56-60	I	INFC(12)	

The set of cards containing the input parameters for the individual materials follows. This entire set is repeated for each material until all materials have been defined.

CARD	COLS.	TYPE	NAME	DESCRIPTION
7	1-5	I	NI	Maximum number of mesh lines allowed in axial direction.
	6-10	I	NJ	Maximum number of mesh lines allowed in radial direction.
	11-15	I	IF	First axial index interior to domain.
	16-20	I	IL	Last axial index interior to domain.
	21-25	I	JF	First radial index interior to domain. Must be 1 if domain touches axis and greater than 1 otherwise.
	26-30	I	JL	Last radial index interior to domain. At least one mesh line must be exterior to the material on all sides (except at axis). Additional exterior mesh lines should be allowed in the directions that the material will move. If the material reaches the end of the allowed mesh in one direction and there is exterior mesh in the opposite direction, it will be automatically shifted.
	31-35	I	NLF	Number of points in first boundary segment.
	36-40	I	NTP	Number of points in second boundary segment.
	41-45	I	NRT	Number of points in third boundary segment.
	46-50	I	NFMAX	Maximum number of boundary points allowed.
	51-55	I	NTR	Number of tracer particles.
	56-60	I	MATNO	Material property number. If positive equation of state constants set in program, if negative they are input.
	61-65	I	IPRINT	Maximum number of interior points printed in axial direction. If positive count from leftmost interior point, if negative count from rightmost interior point.
8	66-70	I	JPRINT	Maximum number of interior points printed in radial direction. If positive count from lowest interior point, if negative count from highest interior point. If either IPRINT or JPRINT is zero no interior points will be printed.
	1-10	R	ZLEN	Initial axial length in centimeters.
	11-20	R	YMAX	Initial radial length in centimeters.
	21-30	R	THETA	89.5
	31-40	R	CAP	1.0E4
	41-50	R	D	Radial length of densest mesh spacing. Radial mesh will be fairly uniform for this length and will then increase in size.
	51-60	R	VIS	Not used.
	61-70	R	VIST	Viscosity coefficient for two-step equations. Should be between zero and two.
	71-80	R	CTL	Fraction of computed time step used. A safety factor usually between 0.5 and 0.8.

CARD	COLS.	TYPE	NAME	DESCRIPTION
9	1-10	R	MU	Shear modulus in 10^{12} dynes/cm ² .
	11-20	R	YC	Yield strength in tension in 10^{12} dynes/cm ² .
	21-30	R	RHO	Initial density in grams/cm ³ .
	31-40	R	UI	Initial axial velocity in centimeters/microsecond.
	41-50	R	VI	Initial radial velocity in centimeters/microsecond.
10	1-10	R	APL	Gruneisen ratio in equation of state.
	11-20	R	BPL	C - dynes/cm ²
	21-30	R	APB	D - dynes/cm ²
	31-40	R	BPB	E - dynes/cm ²
	41-50	R		Not used
	51-60	R		Not used
	61-70	R		Not used
	71-80	R		Not used
	1-10	R		Not used
	11-20	R	PMIN	Maximum tensile stress permitted, dynes/cm ² .
11	1-10	R	TSR(1)	Growth constant - cm ² /dyn/sec.
	11-20	R	TSR(2)	Growth threshold - dyn/cm ² .
	21-30	R	TSR(3)	Nucleation radius parameter - cm.
	31-40	R	TSR(4)	Parameter in nucleation function - no./cm ³ /sec.
	41-50	R	TSR(5)	Nucleation threshold - dyn/cm ² .
	51-60	R	TSR(6)	Parameter in nucleation function - dyn/cm ² .
	61-70	R	TSR(7)	Not used.
	71-80	R	TSR(8)	Threshold stress for entering EFRACT.
	1-10	R	TSR(9)	0 for stress, 1 for deviator stress
	11-20	R	TSR(10)	Ratio of number of fragments to number of cracks.
11a	21-30	R	TSR(11)	Ratio of fragment radius to crack radius.
	31-40	R	TSR(12)	Value of crack volume which defines threshold of coalescence.
	41-50	R	TSR(13)	TF, where fragment volume = TF(RF) ³
	51-60	R	TSR(14)	Not used.
			TSR(8) - TSR(14)	may be left blank for ductile fracture.

CARD	COLS.	TYPE	NAME	DESCRIPTION
12	1-10	R	FRX(I)	Axial coordinate in centimeters of first point of boundary.
	11-20	R	FRY(I)	Radial coordinate in centimeters of first point of boundary.
	21-30	R	FRX(I)	Axial coordinate of second point.
	31-40	R	FRY(I)	Radial coordinate of second point.
	41-50	R	FRX(I)	
	51-60	R	FRY(I)	This sequence is repeated, four points per card, until the entire boundary has been input.
13	61-70	R	FRX(I)	If the boundary touches the axis of symmetry, the defining points must start and end on the axis of symmetry. If the boundary does not touch the axis, the first and last points should be the same and should be on a part of the boundary that will never become a material interface.
	71-80	R	FRY(I)	
				Index of first boundary point selected as tracer.
	1-5	I	ITR	Index of second boundary point selected.
	6-10	I	ITR	
	11-15	I		
	16-20	I		
	21-25	I		This sequence is repeated, sixteen indices per card until all tracer particles are selected. The boundary point indices refer to the points defining the material boundary in the order in which they are input.
	25-30	I		
	31-35	I		
	36-40	I		These tracer points will be tagged with their input number on this card and their position in time will be traced throughout the run.
	41-45	I		
	46-50	I		
	51-55	I		
	56-60	I		
	61-65	I		
	66-70	I		
	71-75	I		
	76-80	I		

IV. DUCTILE FRACTURE RESULTS

A tapered flyer plate, Aluminum-1145, impacting on a flat plate, Aluminum-1145, at a striking velocity $V_g = 0.0163$ cm/ μ -sec. was chosen as a test case. It is, except for the cylindrical geometry, similar to the problem run by SRI which, in that case, was specified in plane coordinates. The coordinate mesh chosen for the computation presented here is not the finest possible but is similar to the SRI mesh.

The target had a mesh spacing $\Delta z = L/20$ where the thickness $L = 0.316$ cm. The spacing of the mesh in the radial direction $\Delta r = h/40$ where the height of the target is $h = 1.184$ cm. In the tapered flyer plate the mesh spacing in z , $\Delta z = 0.158$ cm./12; the radial mesh increment is $\Delta r = 0.592$ cm./24.

The following table gives the parameters (data card 10b) used in the nucleation and fracture model:

FRACTURE PARAMETERS: Aluminum-1145

Growth Constant	-0.01
Growth Threshold	$-0.4 \cdot 10^{10}$
Nucleation Radius	$0.1 \cdot 10^3$
N_0	$0.3 \cdot 10^{10}$
Nucleation Threshold	$-0.3 \cdot 10^{10}$
P_1	$-0.4 \cdot 10^9$

The results of the computation for the value of the parameters given in the above table is presented in the three accompanying figures. Figure 1 shows the geometrical configuration at the moment of impact, $t=0$. At $t=1.464$ μ sec., shown in Figure 2, the target shows the presence of an internal fracture centered about the center line of the plate. This fractured region represents a percent fracture volume between 5 to 12 percent.

The presence of this fracture island is similar to that obtained by SRI, however their calculation was carried out with the brittle model for a flyer plate/target configuration with the material properties of Armco iron. As a result, due to the higher growth rates of this brittle material, the fracture island was of much greater extent, a reasonable result (see the Table below). The last figure shows, at $t=2.198\mu\text{-sec.}$, that there is virtually no continued growth of the fracture island.

FRACTURE PARAMETERS: Armco Iron

Growth Constant	$-0.12 \cdot 10^{-3}$
Growth Threshold	0.
Nucleation Radius	$0.4 \cdot 10^{-2}$
$\dot{N}_0$	$0.5 \cdot 10^9$
Nucleation Threshold	$1.12 \cdot 10^{11}$
P_1	$-7.4 \cdot 10^9$

The brittle fracture routine was substituted for the ductile fracture routine, and the axisymmetric flyer plate problem was run with Armco iron properties rather than Aluminum-1145 properties. At this point, we have very limited results. The main difficulty is a convergence problem in the brittle fracture subroutine. We have been conversing with the scientific staff at SRI who are responsible for the brittle fracture model. As a result of their suggestions, there has been improvement in the convergence of the iterative procedure in BFRACF but not a complete solution to the problem.

DUCTILE FRACTURE-AL 1145

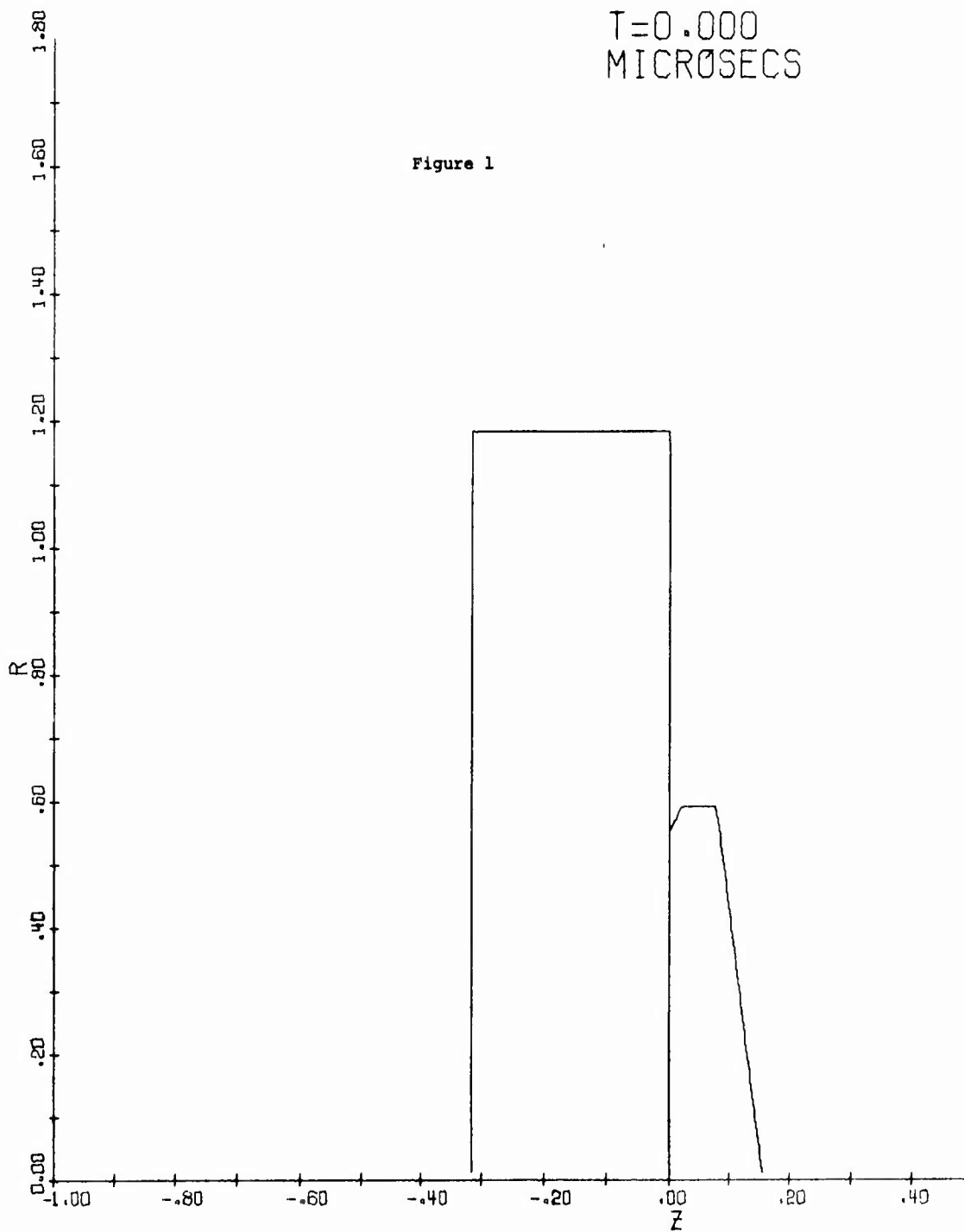
6.0	1.46	0.73	0.73	-0.73	1500.0	-1.0	0.25
0.5							
2	0						
1							
2	1	2	1	5	4		
23	41	2	21	1	40	2	2
0.3156	1.184	89.5	1000.0	1.184	0.0	-1	999
0.3	0.002	2.7	0.0	0.0	0.0	0.0	0.65
2.04	0.76	1.5	0.0	0.0	0.0	0.0	0.0
0.0	-9.999						
-1.0	E-2 -4.0	E9 1.0	E-4 3.0	E9 -3.0	E9 -4.0	E8 0.0	
-0.3156	0.0	-0.3156	1.184	0.0	1.184	0.0	0.55
0.0	0.0						
1							
16	29	4	15	1	24	2	3
0.1578	0.592	89.5	1000.0	0.592	0.0	-2	999
0.3	0.002	2.7	-0.0163		0.0	0.0	0.65
2.04	0.76	1.5	0.0	0.0	0.0	0.0	0.0
0.0	-9.999						
-1.0	E-2 -4.0	E9 1.0	E-4 3.0	E9 -3.0	E9 -4.0	E8 0.0	
0.0	0.0	0.0	0.55	0.02	0.592	0.0789	0.592
0.1578	0.0						
1							

BRITTLE FRACTURE-STEEL-2151-3

6.0	1.0	1.0	1.0	-2.0	500.0	-1.0	0.2
0.5							
2	0						
1							
2	1	2	1	5	4		
16	31	2	15	1	30	2	3
0.4233	1.8	89.5	1000.0	1.8	0.0	1	-1 999 999
0.819	0.01835	7.85	0.0	0.0	0.0		0.65
1.69	1.589	5.17	51.7	0.0	0.0		0.0
0.0	-9.999						
-1.2	E=4	0.0	4.0	E=3	5.0	E8 -1.12	E10 -7.4
0.0	0.25	1.0	0.2	1.0	0.0	E8 0.0	0.0
-0.4233	0.0	-0.4233	1.8	0.0	1.8	0.0	0.85
0.0	0.0						
1							
23	32	3	22	1	30	2	3
0.3048	0.9	89.5	1000.0	0.9	0.0	1	-1 999 999
0.819	0.01835	7.85	-0.05	0.0	0.0		0.65
1.69	1.589	5.17	51.7	0.0	0.0		0.0
0.0	-9.999						
-1.2	E=4	0.0	4.0	E=3	5.0	E8 -1.12	E10 -7.4
0.0	0.25	1.0	0.2	1.0	0.0	E8 0.0	0.0
0.0	0.0	0.0	0.85	0.01	0.9	0.0762	0.9
0.3048	0.0						
1							

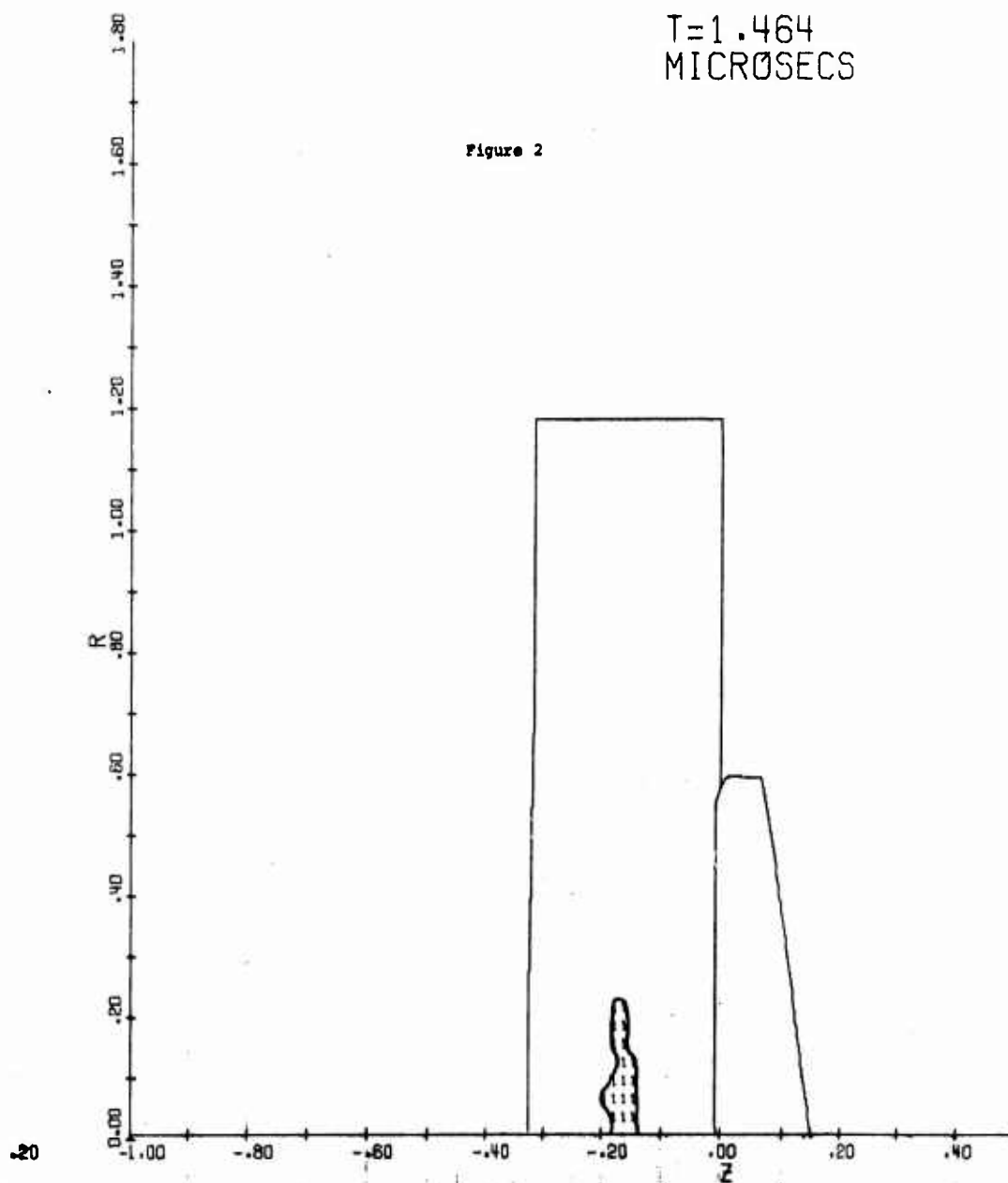
T=0.000
MICROSECS

Figure 1



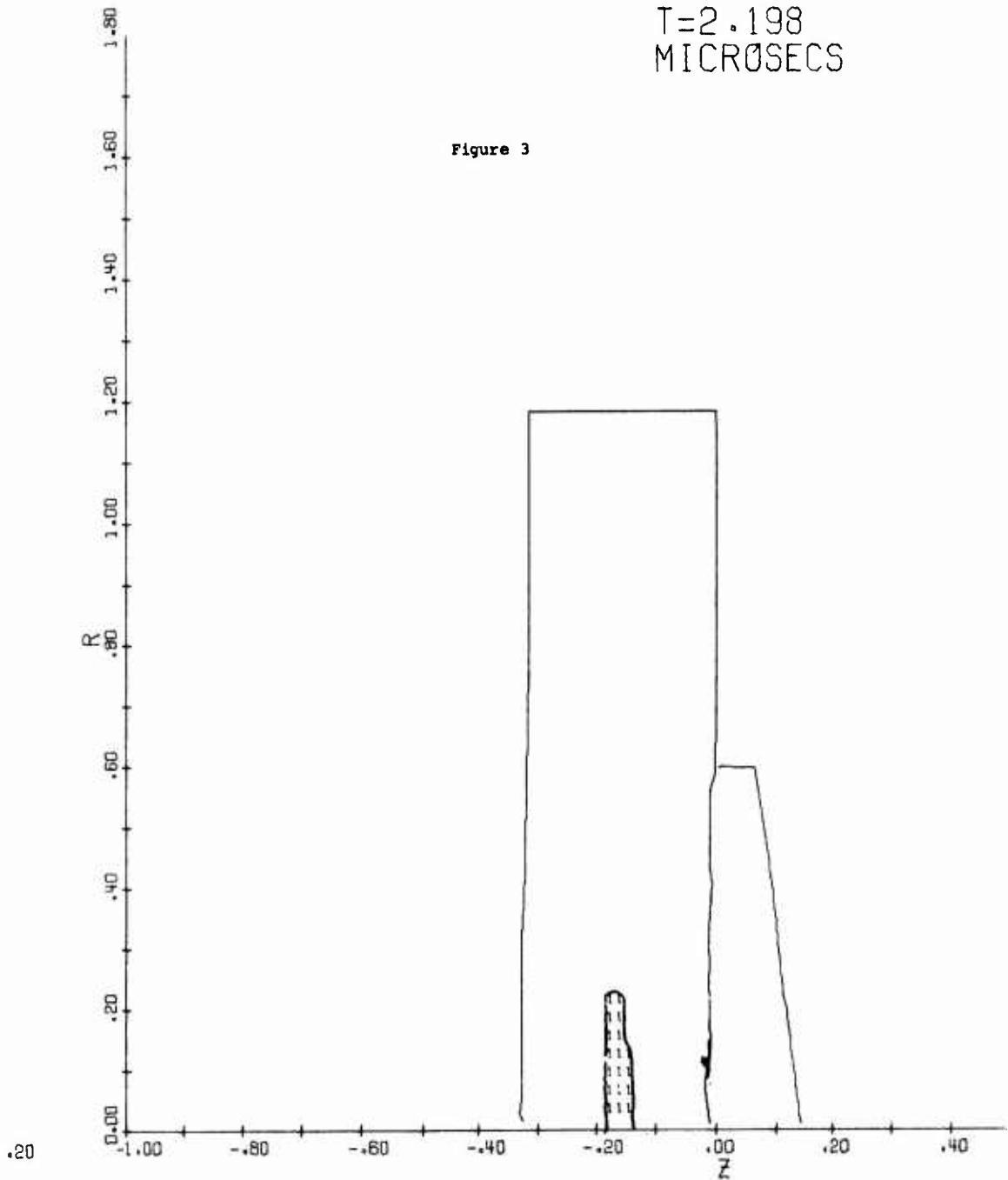
$T=1.464$
MICROSECS

Figure 2



T=2.198
MICROSECS

Figure 3



V. SUMMARY AND RECOMMENDATIONS

The feasibility of adapting the SRI Lagrangian fracture subroutines for inclusion in the SMITE Eulerian code has been demonstrated. The ductile fracture subroutine was successfully run with reasonable results. The brittle fracture subroutine was run, but aborted due to convergence problems. SRI was consulted about the problem and recommended corrections were received. However, the corrections were not received until after the funds under the present contract were expended. Those corrections, therefore, have not been included in the code.

It is recommended that the following additional items be considered by BRL:

- 1) Inclusion and testing of the corrections recommended by SRI with additional consultation, if necessary.
- 2) More extensive testing of both the ductile and brittle fracture subroutines to determine if any further corrections are necessary.
- 3) Further modification of printed and graphic output to present results in a form suitable to the requirements of BRL.

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