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September 1960

DEVELOPMENT OF TITANIUM ALLOY CASTING METHOD

R. V. Carter

AERO-SPACE DIVISION
BOEING AIRPLANE COMPANY
Contract: AF33(600)-36450
AMC Project: 7-656

FINAL TECHNICAL ENGINEERING REPORT

March 1958-May 1960

A commercially feasible process was developed for production of titanium alloy castings for high performance aerospace vehicle applications. The recommended practice utilized a vacuum consumable-electrode arc furnace, a water cooled copper, tilt-pour crucible, expendable rammed graphite molds, and centrifuge casting techniques. The Ti-6Al-4V alloy is the best casting alloy presently available, with as-cast tensile strength in excess of 150,000 psi and excellent surface finish.

METALLIC MATERIALS BRANCH
MANUFACTURING AND MATERIALS TECHNOLOGY DIVISION

AMC Aeronautical Systems Center
United States Air Force
Wright-Patterson Air Force Base, Ohio

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This is the final technical engineering report covering all work performed under Contract AF33(600)-36450 from March 1958 to May 1960. The manuscript was released by the author on May 30, 1960 for publication as an AMC Technical Report.

This contract is a continuation of the work started by National Research Corporation to develop titanium casting fundamentals under Contract AF33(600)-32801, Project 7-216-n.

This contract with Boeing Airplane Company was initiated under AMC Manufacturing Methods project number 7-656, "Development of Titanium Alloy Casting Method." The contract was administered under the direction of Mr. A. H. Langenheim of the Metallic Materials Branch (LMBML-1), Manufacturing and Materials Technology Division, AMC Aeronautical Systems Center, Wright Patterson Air Force Base, Ohio.

Mr. R. V. Carter, Research Engineer, Aero-Space Division of Boeing Airplane Company was the engineer in charge of the program. The work was supervised by Mr. J. W. Sweet, Chief Metallurgist, Materials and Processes Staff, Aero-Space Division, Boeing Airplane Company.

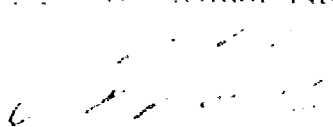
The primary objective of the Air Force Manufacturing Methods Program is to increase producibility and improve quality and efficiency of fabrication of aircraft and missiles and components thereof. This report is being disseminated in order that the methods and/or processes developed may be used throughout industry, thereby reducing production costs and obtaining "MORE AIR FORCE PER DOLLAR".

Your comments are solicited on the potential utilization of the data and information contained in this report as applied to your present or future production program. In addition, any suggestions concerning additional Manufacturing Methods developments required on this or similar subjects will be appreciated.

PUBLICATION REVIEW

This report has been reviewed and is approved.

FOR THE COMMANDER


PRESTON L. MILLER, Major, USAF
The Boeing Company, Aeronautical Systems Center

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The subcontractor for this developmental contract was Oregon Metallurgical Corporation, Albany, Oregon. The author particularly wishes to acknowledge the major contributions made to this program by Messrs. R. G. Hardy, J. A. Smith, T. A. Hamm, F. Caputo, and J. L. Gunter.

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CONTENTS

	Page
A. Introduction	A 1
B. Casting Practice Development	B 1
Melting Practice Development	B 2
Furnace Characteristics and Operation	B 2
Electrode Preparation	B 4
Recycling of Scrap	B 5
Mold Development	B 15
Machined Graphite Molds	B 15
Metal Molds	B 15
Rammed Graphite Molds	B 16
Shell Graphite Molds and Cores	B 20
Investment Molds	B 22
Recycling of Mold Materials	B 24
Feeding Studies	B 30
Feeding Distance in Flat Sections Cast in Machined Graphite Molds	B 30
Feeding Distance in Tapered Sections Cast in Machined and in Rammed Graphite Molds	B 31
Feeding Distance in Shell Graphite Molds	B 33
Feeding Distance in Centrifuge Casting	B 34

	Page
Surface Treatment of Titanium Coatings	B56
Removal of Gases and Kinks	B56
Cleaning of Coatings	B57
Chemical Removal of Surface Material	B58
C. Alloy Development	C 1
Development of Ti-6Al-4V Alloy	C 2
Development of Other Alloys	C 7
D. Casting of Shapes	D 1
Selection of Shapes	D 2
Casting of the Developmental Bracket	D 4
Casting of the Flap Track Link	D26
Casting of the Sway Brace Case	D47
Casting of the Elevator Torsion Fitting	D55
Casting of the Horizontal Stabilizer Fitting	D74
E. Evaluation Testing	E 1
Conventional Materials Tests	E 2
Component Tests	E 4
Developmental Bracket	E 4
Flap Track Link	E10
Sway Brace Case	E13
Elevator Torsion Fitting	E15
Horizontal Stabilizer Rib	E18

	Page
F. Specifications, Standards, and Allowables	F 1
Casting Specification	F 2
Design Allowables	F 7
Reference Radiographs	F10
Dimensional Tolerances	F11
G. Conclusions	G 1
H. Recommendations	H 1
J. Tables	J 1
K. Appendix	K 1

TABLES

	Page
J 1 Melting Record	J 1
J 2 Mechanical Properties of Recycled Low Interstitial 6Al-4V	J19
J 3 Experimental Rammed Graphite Mold Mixtures	J21
J 4 Grain Size Distributions of Graphite Base Materials	J25
J 5 Properties of Rammed Graphite Mold Mixtures	J26
J 6 Shrinkage of Rammed Graphite Molds	J33
J 7 Experimental Shell Graphite Mold Mixtures	J34
J 8 Experimental Investment Mold Mixtures	J35
J 9 Grain Size Distributions of Recycled Mold Materials	J36
J10 Feeding Distances in Un-tapered Machined Graphite Molds	J37
J11 Feeding Distances in Tapered Machined Graphite Molds	J38
J12 Feeding Distance in Tapered Rammed Graphite Molds	J42
J13 Feeding Distances in Heated Machined Graphite Molds	J47
J14 Feeding Distances in Heated Rammed Graphite Molds	J50
J15 Feeding Distances in Centrifuged Discs	J53
J16 Results of Tension Fatigue Tests	J54
J17 Mechanical Properties of As-Cast Ti-6Al-4V	J55
J18 Elevated Temperature Properties of As-Cast Ti-6Al-4V	J60
J19 Mechanical Properties of Heat Treated Cast Ti-6Al-4V	J61
J20 Mechanical Properties of As-Cast Experimental Alloys	J64

		Page
J21	Mechanical Properties of Heat Treated Experimental Alloys	J71
J22	Heat Analyses	J80
J23	Results of Developmental Bracket Fatigue Tests	J90
J24	Results of Statistical Evaluation of Dimensions	J91

SECTION A

INTRODUCTION

This is the final report on Contract AF 33(600)-36450. This research and development contract was initiated approximately 1 March 1958 and had the following objectives:

- (a) Develop a complete commercially feasible process, on a non-proprietary basis, capable of producing close-tolerance titanium-alloy castings of the type, size, and quality required for aircraft application.
- (b) As a practical demonstration of the validity of the developed process, produce and evaluate pilot production quantities of several different casting designs.
- (c) Establish procurement specifications, quality control procedures, inspection methods and standards, and design criteria necessary for extensive application of titanium designs.

This program consisted of four phases summarized as follows:

- | | |
|---------------------------------|--|
| Phase I - Preliminary. | Establish the present state of the art; select components for development as titanium castings. |
| Phase II - Process Development. | Develop a practicable casting process; investigate melting and pouring procedures, mold materials, mold design, casting alloys, surface treatments, and casting design. |
| Phase III - Trial Production. | Produce and evaluate test parts; develop quality control methods, inspection standards, procurement specifications, design procedures, heat treatment procedures; establish design allowables by test. |

Phase IV - Pilot Production.

Produce pilot production lots of several casting designs in accordance with specifications, etc. developed in Phase III.

Oregon Metallurgical Corporation was selected as subcontractor for this program, and conducted the necessary foundry research and development work.

SECTION B

CASTING PROCESS DEVELOPMENT

MELTING PRACTICE DEVELOPMENT

Because of the extreme reactivity of titanium at and above its melting point, an entirely new melting process has been developed specifically for casting titanium and similar metals. The rapid contamination of titanium at elevated temperature by oxygen and nitrogen and the severe damage to ductility by small percentages of such contaminants requires that all high temperature processing of titanium be done in the absence of air. Also, since molten titanium reacts readily with almost all crucible materials, the problem of containing the heat until enough stock is melted to make a pour has required special attention. The only satisfactory furnace for production casting of titanium was developed previous to this program by the Albany, Oregon experimental station of the U.S. Bureau of Mines. The equipment used in this program is patterned after the Bureau of Mines furnace, with improvements in the mechanisms and controls. Briefly, the furnace consists of a cylindrical tank with a tower for containing and feeding the electrode, a water cooled copper crucible supported on trunnions inside the tank, provisions for centrifuge casting, and the necessary vacuum pumps and controls. A sketch of the vertical axis centrifuge furnace is presented as Figure B1.

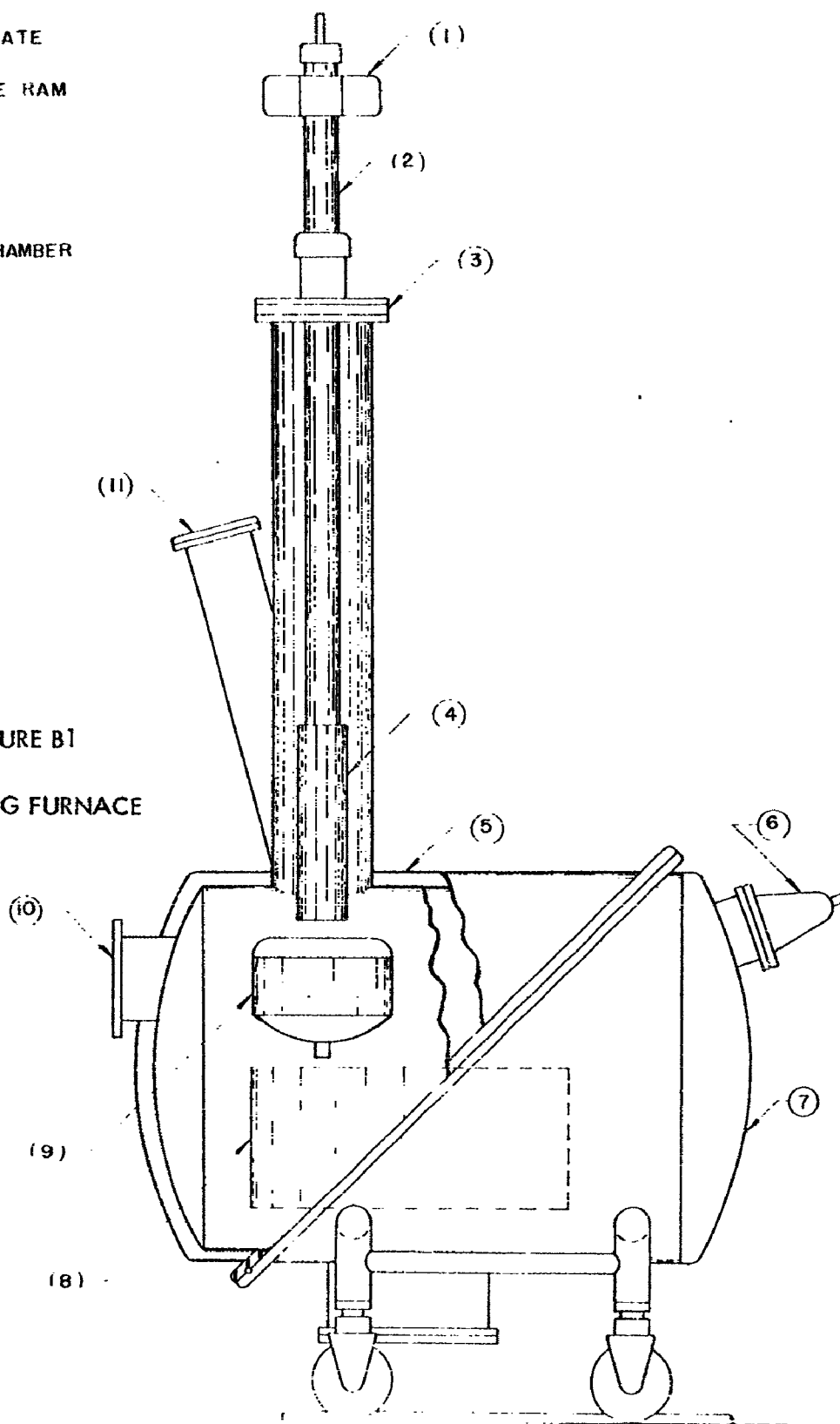
Furnace Characteristics and Operation

Two furnaces were used during this program. One was a static casting furnace with provisions for horizontal axis centrifugal casting. The second was a vertical axis centrifugal casting furnace designed and built by the subcontractor to meet requirements developed during this program. Both units operate with the electrode at negative polarity.

All experimental melting and casting was done under a dynamic vacuum. Melting was started by striking an arc between the electrode and a small quantity of sponge or solids placed in the copper crucible or against the skull from a previous melt. Melting power was then rapidly increased to the desired level where it maintained until the electrode was consumed and the molten pool adequately superheated. Upon accumulation of the proper amount of molten metal in the crucible, the power is cut, the electrode is

1. POWER LEAD CLAMPING PLATE
2. WATER COOLED ELECTRODE RAM
3. MICARTA INSULATOR
4. ELECTRODE
5. WATER COOLED VACUUM CHAMBER
6. LIGHT PORT
7. MOBILE VACUUM CHAMBER
8. MOLD SPIN TUB
9. WATER COOLED CRUCIBLE
10. VACUUM PORT
11. VIEW PORT

FIGURE B1
CASTING FURNACE



rapidly retracted by an pneumatic cylinder, and the crucible tipped to make the pour. Pouring of the mold is completed in four to six seconds after the arc is extinguished. Table J1 lists the melting conditions recorded for each heat made in this program. The maximum poured weight during this program was 340 pounds.

Cooling of the molds and furnace is usually accomplished by backfilling the furnace chamber with inert gas, or occasionally by allowing slow cool in vacuum.

Electrode Preparation

Electrodes for melting were prepared from ingot, wrought products scrap, or casting recycle material such as gates, risers, and rejected castings. An electrode prepared from ingot requires only the welding of the stub (which is attached to the stinger) to the ingot. The preparation of an electrode from scrap as shown in Figure B2 requires additional fabrication of the electrode by welding.

The general procedure for preparation of an electrode from casting recycle material is as follows. The gates and risers are removed from the castings by power saw, abrasive cut-off, or by oxy-acetylene torch. The material is then cleaned by shot blast. The scrap is segregated to permit control of chemical analysis by blending. The cleaned material is next assembled into an electrode by Heliarc welding in air atmosphere. Figure B2 is a photograph of an electrode of commercially pure titanium which was used to pour experimental castings. This electrode was made from sprues, gates, and risers from previous heats plus a slab of wrought scrap plate which was added to dilute oxygen to an acceptable level. The entire electrode, except the attachment stub, is melted in one pour.

The welds must be large, since they must carry the entire current load being used to melt the electrode. In some instances, inadequate welds have resulted in electrodes being prematurely dropped into the crucible. As the consumable electrode is melted, hot spots occasionally occur on the electrode, requiring that the power be reduced. If welding cracking is a problem in fabrication of alloy electrodes, commercially pure weld filler wire can be used to dilute the alloy content in the weld. The electrodes are welded as symmetrically as possible to avoid protrusions which could arc to the crucible.

Recycling of Foundry Scrap

It was decided that maximum recycle of gates, risers, and other foundry scrap would be utilized to gain experience on the effects of melting on chemical analysis and the effects of analysis on mechanical properties.

Three series of heats were poured to experimentally determine the sources of melting contamination of the cast titanium alloy. The variables in each series were as follows:

Series 1 - Control of melting at the normal operating standards, with electrode preparation by welding in an inert-gas (helium) tank using carbon welding electrode.

Series 2 - Melt under varying furnace vacuum (simulating poor melting practice), with electrode preparation as in Series 1.

Series 3 - Controlled melting as in Series 1, with electrode preparation by cutting and welding (Heliarc) in air atmosphere.

The results of these trials are shown in the series of graphs comparing each of three series of casting conditions. Each of the three casting conditions are plotted on the same graph for comparing the interstitial rise in each controlled series.

Figure B3 illustrates the vacuum control held during the three casting series. The three curves plotted on each graph (Series 1, 2, and 3) are drawn to show the starting vacuum reading, the vacuum reading during the melting cycle, and the vacuum reading directly after pouring. The vacuum is measured in microns of pressure contained in the furnace chamber. It is evident from the "during" melting curve that outgassing occurs from the melting of the ingot and as continued recycling is conducted, the amount of outgassing increases. The "after" pouring reading follows very closely the "during" melting vacuum readings, demonstrating that little gas is liberated by pouring metal into the graphite mold.

The vacuum pressure in Series 2 deliberately varies over a large range as this series was conducted to determine the effect of varied melting conditions. The first recycle heat of the Series 2 tests was started at 50 microns and the remaining heats were varied to complete the recycle experiment. The conditions in the furnace were varied by backfilling to maintain the desired vacuum reading. The second recycle heat in Series 2 illustrates that although the furnace was backfilled to 200 microns, vacuum condition improved during the run due to the continuous pumping of the furnace atmosphere.

The casting trials in Series 3 were conducted to determine the effect of welding the electrodes in air. As the vacuum curve shows, control of atmosphere during the entire series closely parallels Series 1, illustrating that normal vacuum control was held. The "during" and "after" vacuum readings vary over a wide range relative to the initial micron readings, due to welding of the electrodes in air. Because of the method in which the electrode was made for the continuous recycle heats, little welding was necessary. To determine the effect of conventional welding requirements on interstitial contamination, additional welding was performed by welding beads along the side of the ingot. It is evident from the graphs that outgassing occurred during the melting stage in larger proportion than in the other two series. It was concluded that the welding in air does increase gas evolution during melting.

The series of graphs in Figure B4 illustrate the increase in interstitials in the recycled 6Al-4V. The graphs show carbon, oxygen, hydrogen, nitrogen, and iron content. Each of the series of casting conditions plotted on the same graph illustrates by comparison, the increases in interstitials due to different melting conditions.

The carbon percentage graph demonstrates that in the original heat, carbon increased as the recycle heats continued. Carbon in recycle heat five is low, probably because that electrode was pickled prior to casting. In recycle heat six the carbon content again increases. It is evident that the increase in carbon is a function of the number of times the material is cast in the graphite mold. Melting conditions in Series 2 evidently did not have a separate effect upon carbon increase. The rate of carbon increase is similar to that of Series 1.

The increase in oxygen does not show a great deal of variation, there being a slight increase in all the heats as recycling continued. Poor atmosphere control and welding in air are both detrimental to oxygen control. Since oxygen must be maintained within close limits, the foundry has the option of tight process controls to minimize oxygen pickup during the process cycle or addition of higher purity raw stock to the heat to dilute the contamination to the proper level.

The hydrogen curve illustrates a rapid increase in hydrogen when electrode is fabricated in air by welding with Hellarc.

The increase in nitrogen occurs slowly as recycling continued. In heat five of Series 1, a reduction in nitrogen is noted on the graph. The reason is not known. Series 2 increases in nitrogen are gradual and uniform and increases in nitrogen, therefore, seem to be consistent with uncontrolled furnace conditions.

The increase in iron in recycling titanium does not seem to be separately influenced by furnace conditions. It is expected that the iron content would not increase appreciably as recycling continued. The source of iron contamination was shot blasting of the ingot material for cleaning and, as very little cleaning was necessary for recycling, the contamination level was kept quite low.

Mechanical property tests of each of the recycled heats are reported in Table J2 and are plotted on graphs in Figure B5. Yield strength, ultimate tensile strength, elongation, reduction of area, and Brinell hardness are included. Each plotted point is based on the average of three tensile specimens tested. Series 1 strength curves show a small increase in strength over the period of recycle heats. The second series strength curves show higher, but consistent strength in recycle heats except for heat four, where the specimens failed without yield.

The Series 3 conditions resulted in a gradual increase in strength as recycling continued.

The ultimate strength curves are generally similar to those shown for yield strengths. The average ultimate tensile strength of the fourth heat of Series 2 dropped to 81,000 psi, with no ductility.

After recycle heat three, the chemical analyses and properties showed greatest variation between the three practices investigated. This indicates that recycling of selected foundry scrap is acceptable if the scrap is properly diluted with new metal.

FIGURE B2

TYPICAL ELECTRODE



FIGURE B3

FURNACE PRESSURES DURING EXPERIMENTAL RECYCLE TRIALS

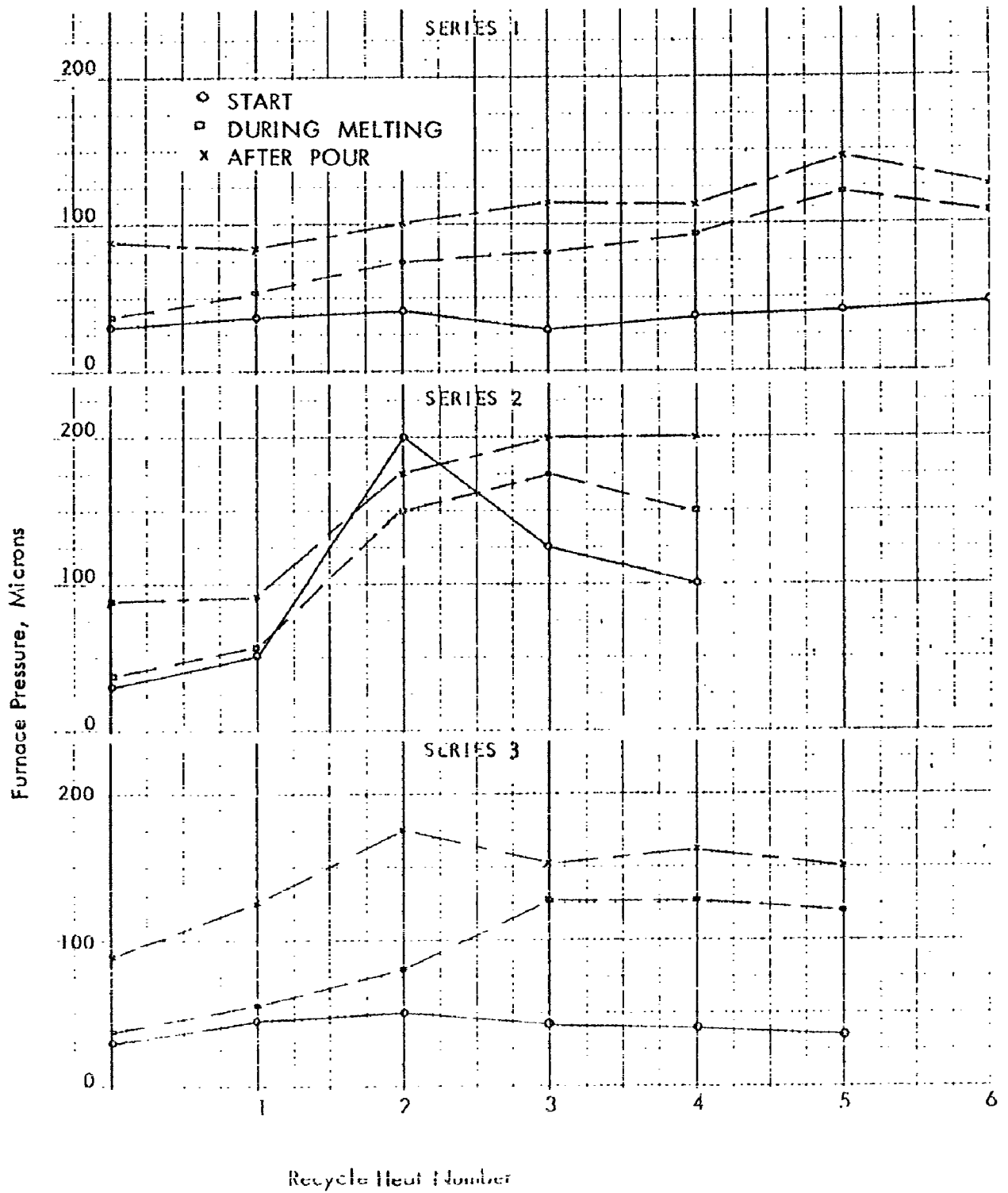


FIGURE B4

INTERSTITIAL ANALYSES OF RECYCLE HEATS

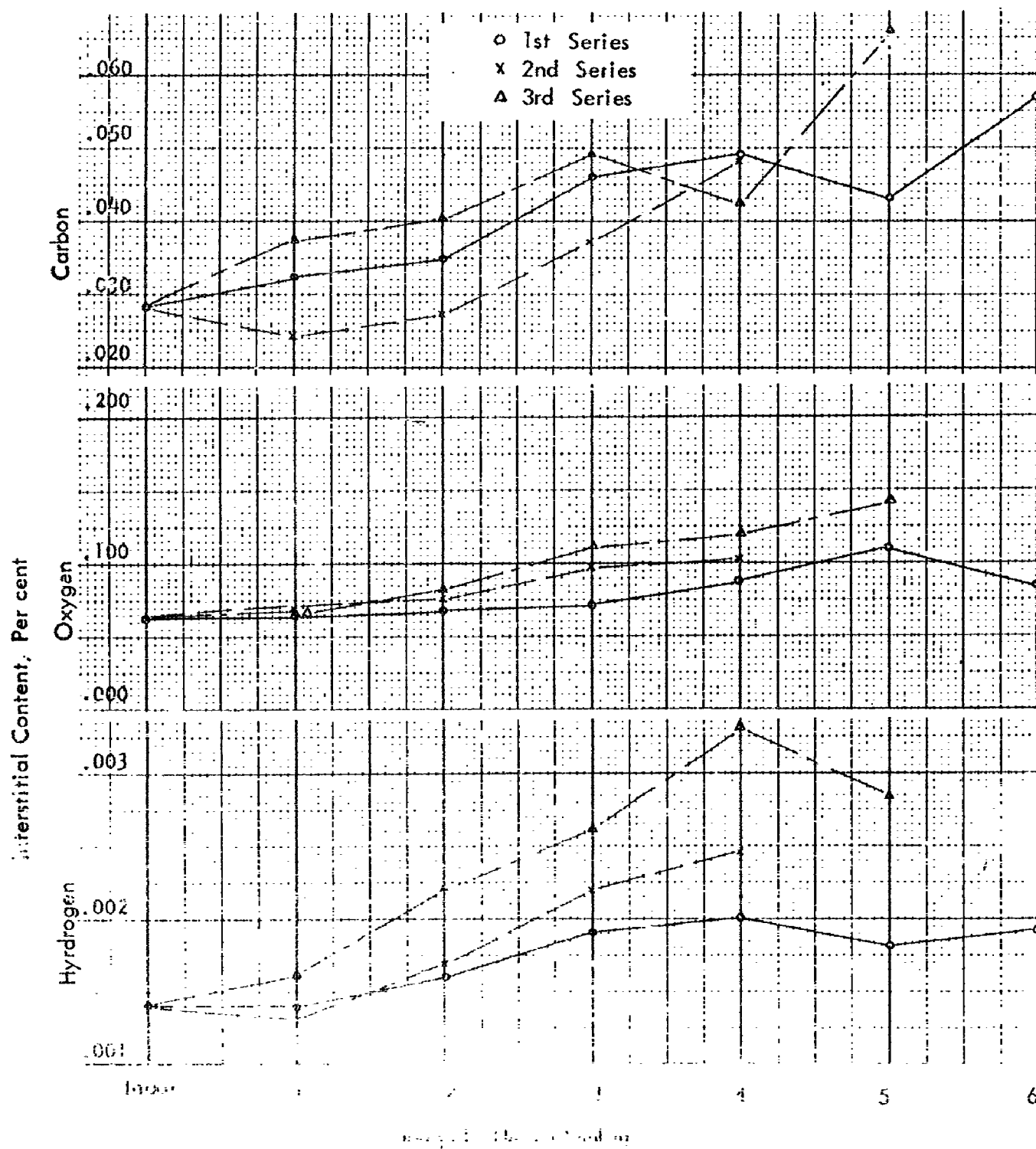


FIGURE B4 (Continued)

INTERSTITIAL ANALYSES OF RECYCLE HEATS

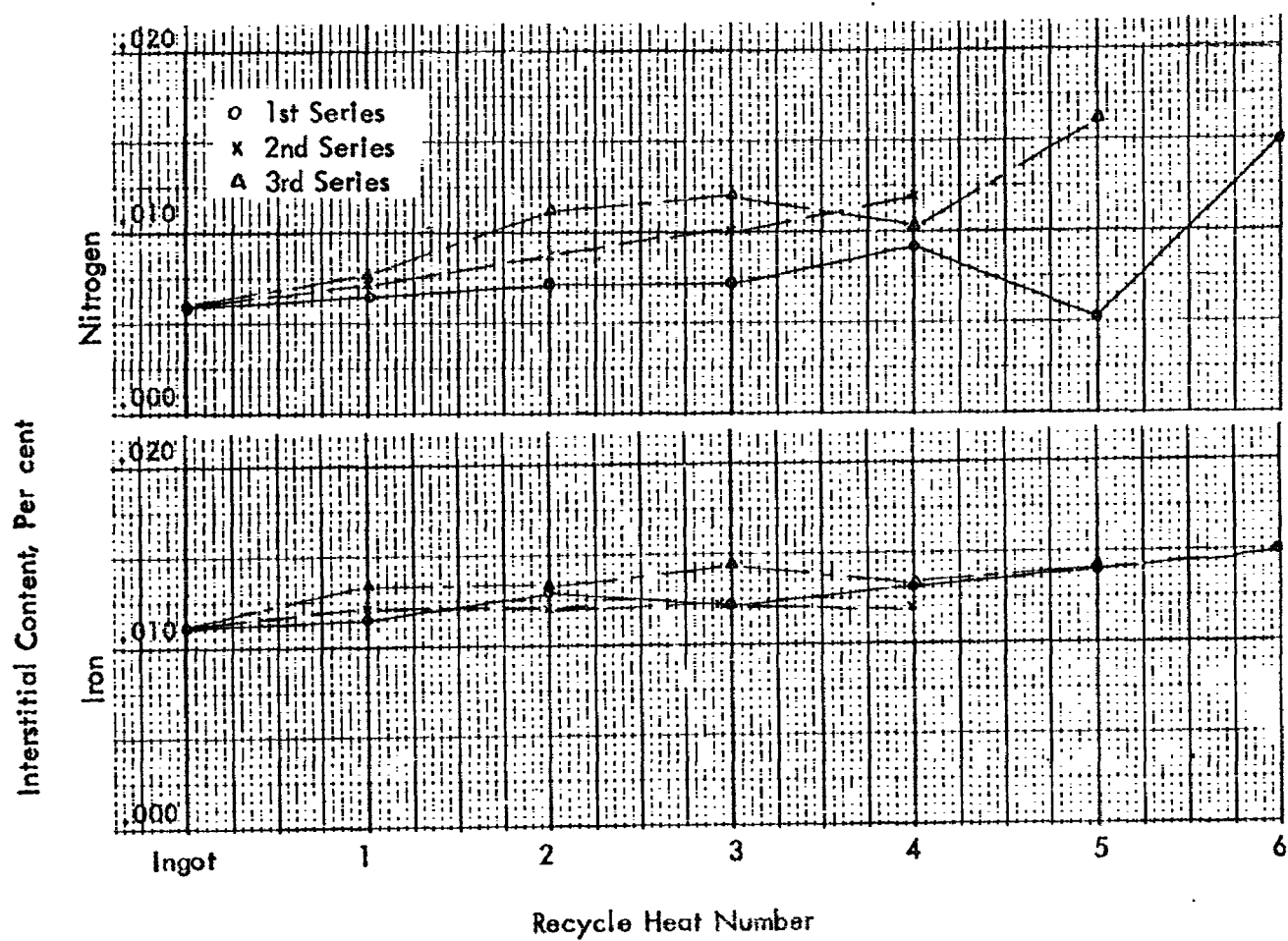


FIGURE B5

PROPERTIES OF RECYCLED Ti-6Al-4V

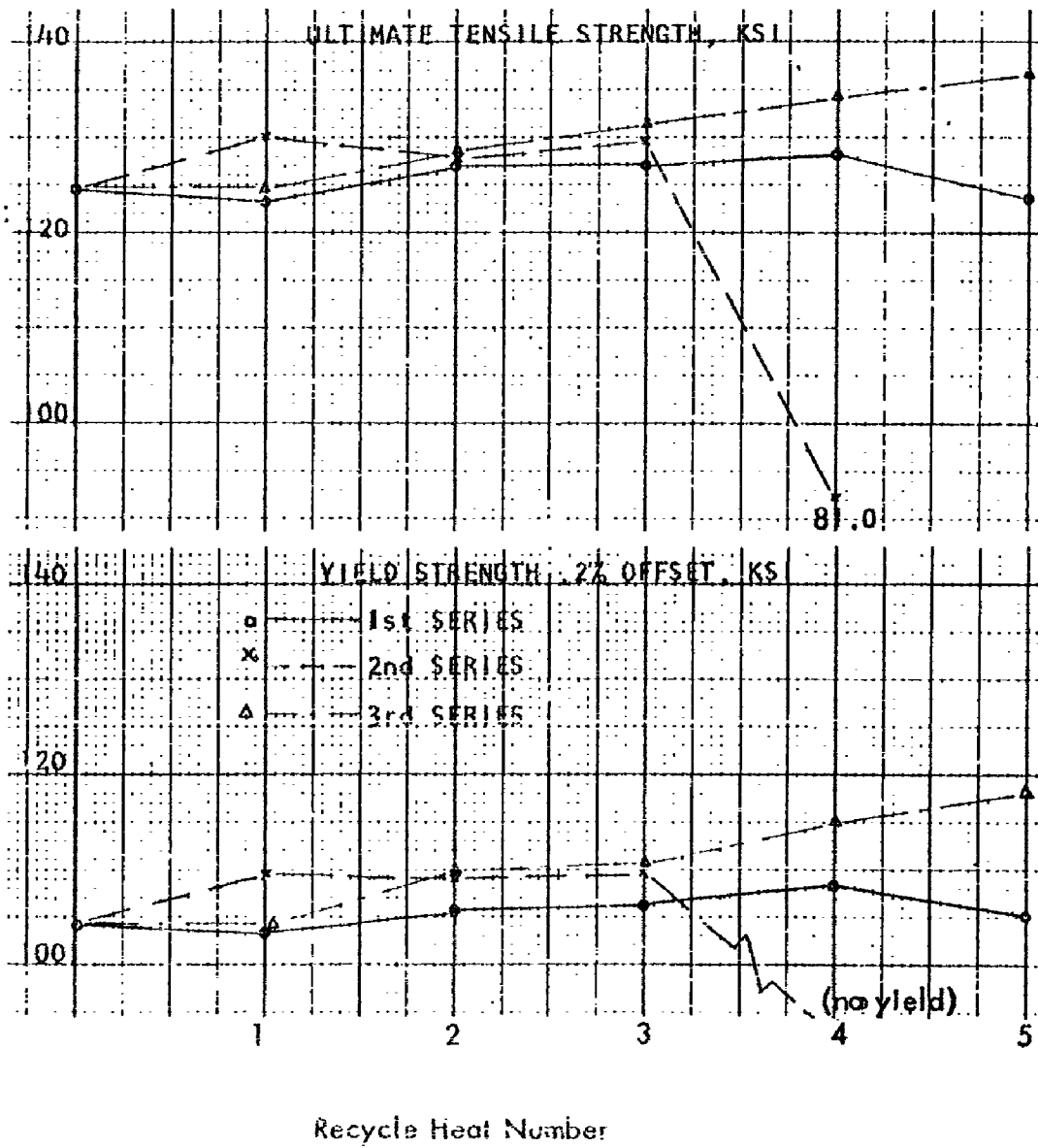
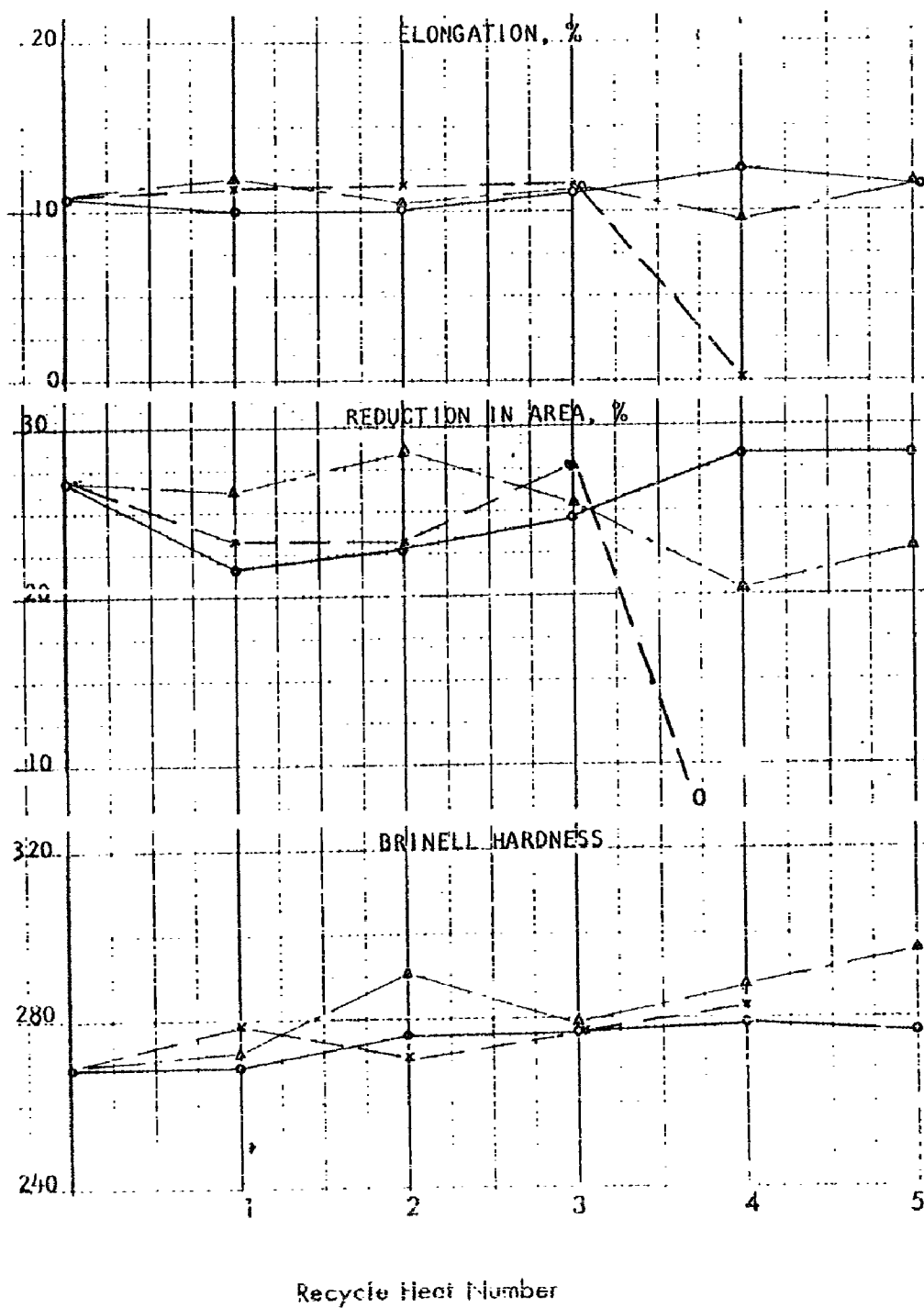


FIGURE B5 (Continued)

PROPERTIES OF RECYCLED Ti-6Al-4V



MOLD DEVELOPMENT

A major problem in the development of a practical process for casting of the reactive metals has been that of finding a suitable mold. The conventional mold materials used for casting of steels (such as sand, shell, or ceramic molds) are completely unsatisfactory for titanium, since titanium above or near its melting point will violently react with those materials. The only materials which have been found to satisfactorily withstand contact with molten titanium are metals or graphite. Of these, graphite has been the most satisfactory.

Machined Graphite Molds

At the time of initiation of this contract, usable titanium castings of simple geometry had been produced in machined graphite molds. This material was used for much of the early work in this program, such as the first trial casting of the developmental bracket and the first feeding studies. It was soon found to be unsatisfactory for the developmental bracket, because the high strength of the mold would not allow proper contraction of the casting during cooling, and consequently would cause hot tearing in the restricted web area and mold spalling, particularly at corners and restricted areas. Mold life can be extended by use of replaceable mold inserts.

In general, machined graphite molds are useful for simple shapes, where the mold does not restrict contraction of the metal during cooling. Reproduction of surface detail is good, and close dimensional tolerances can be maintained. Some loss of dimensional tolerance capability occurs if mold inserts are used.

Metal Molds

Metal molds are satisfactory for only the simplest of shapes. Any restriction of contraction during cooling of the casting will cause rupturing of the restricted areas. The cost of preparing a metal mold is high.

Rammed Graphite Molds

Shortly before initiation of this program, details were published on research work on rammed graphite mold mixtures carried out by A.L. Feild, Jr. at the E.I. du Pont de Nemours experimental station at Wilmington, Delaware. The mold mixture consisted of powder graphite as a base, corn starch for a green binder, pitch and carbonaceous cement for a high-temperature binder, and water with a wetting agent to provide molding plasticity. Further development of the mold was done by Andes, Norton, and Edelman at Frankford Arsenal. This work provided a starting point for development of a suitable rammed-graphite mold for production casting of titanium.

A total of 133 experimental mold mixtures were prepared and tested during this program. Parameters used in judging the mixtures were mold shrinkage; presence or lack of reaction with titanium; carbon contamination of the titanium; green strength, permeability, and hardness; and general handling and casting characteristics.

All batches were mixed in a Carver Muller, Model I-GF. The general mixing procedure was:

- (a) mull graphite (or other base) and starch dry for two minutes,
- (b) add pitch and mull for two minutes,
- (c) add water and cement and mull for four minutes.

Samples of each batch were hand screened through an eight-mesh riddle for testing. In most cases, molds were prepared from the remaining material for shrinkage determinations and casting trials. Compositions of the various mixes are in Table J3. Grain size distributions of the graphite base materials are in Table J4.

Two-inch diameter by two-inch length standard American Foundrymen Society briquette specimens and standard A.F.S. tensile specimens were prepared from the screened material. These specimens were hand rammed to a hardness of A.F.S. 80 when mix properties permitted.

After the properties of the green specimens had been determined, specimens were air dried for a minimum of 16 hours at 250°F and then fired. Firing was accomplished by packing the specimens in graphite powder in corrosion resistant steel boxes, and placing the box in an electric-resistance heated furnace at 1200°F. The temperature of the furnace was raised to 1600°F for three hours and the furnace then turned off and allowed to slowly cool. The specimens were removed after the furnace had cooled to 1200°F. After air cooling to room temperature, the specimens were tested using standard foundry sand-testing equipment. The results of tests on the trial mold mixtures are given in Table J5.

Next, rammed molds for casting 5/8-inch thickness by six-inch diameter plates were prepared from the experimental mixtures. These molds were used to determine mold shrinkage during firing and to determine carbon pickup, resistance of the mold to reaction with the titanium, and surface quality of the cast part. Most of the molds were produced using a split core box technique where the pattern halves could be assembled to form the completed mold. Molds were bench rammed to a mold hardness of A.F.S. 80 using a pneumatic hand rammer. The patterns were drawn using a vibrator draw machine.

The molds were partially dried at room temperature and finish dried at 250°F, usually for 16 hours. They were then packed in graphite powder in corrosion resistant steel boxes and placed in an electric-resistance heated furnace at 1200°F. The furnace was then raised to 1600°F, held for three hours, and the power shut off. The molds were removed when the furnace had cooled to 1200°F. The firing cycle required approximately 24 hours. Fired molds were stored at 250°F until about two hours before pouring, when they were assembled. It is desirable to keep the molds hot until they are placed into the casting chamber, to avoid moisture pickup from the atmosphere. This is difficult since the external gating and the mold halves and cores must be manually assembled after the mold is fired. Later in the program, molds for production castings were heated to 1650°F instead of 1600°F, in an attempt to reduce gas porosity defects.

The results of hardness tests on the experimental rammed molds are in Table J5. The shrinkage measurements (made by measuring outside dimensions of molds before and after firing) are given in Table J6.

Carbon contamination adjacent to the surface of Ti-6Al-4V castings produced in rammed-graphite and machined-graphite molds was determined by machining 0.003-inch successive cuts from the face of cast discs and analyzing the cuttings obtained. The edges of the discs had been previously machined so that contamination at the edge would not influence the analyses. Results of these tests are plotted in Figure B6. As shown by this graph, the carbon contamination is high adjacent to the surface, but decreases rapidly as depth increases, and at .025 to .030-inch has leveled off to the base carbon content of the casting. This depth of contamination was confirmed by hardness traverse technique.

Mixtures 7 through 15 were produced to evaluate binders. Observations of interest are that laundry starch provides fired strength, pitch provides fair green strength and best fired strength, carbonaceous cement contributes only to baked strength, and foundry corn flour provides good baked strength but is weak after firing. Starch-pitch and starch-cement combinations provide good green and fired properties.

Mixtures 19 through 29 evaluate the use of raw linseed oil as a substitute for carbonaceous cement. Three and four percent additions of linseed oil produced good molds without detriment to castings poured in these mixtures. However, the mixtures containing linseed oil were difficult to handle because of "stickiness". Costs of the two materials were found to be about equal.

Mixtures 34 through 38 were made with varying amounts of carbonaceous cement and with corn flour as a substitute for laundry starch. Although the tests of the low cement mixtures indicated desirable properties, the molds tended to be weak and broke easily at corners.

Mixtures 39 through 133 were made to investigate substitute materials such as coke for graphite, to test a wetting agent, and to find mixtures that would decrease mold shrinkage.

The addition of "DuPont G" (du Pont), a wetting agent, to the rammed graphite mold mixture caused significant increase in fired tensile strength, but caused stickiness of the mixture making preparation of molds more difficult.

The use of conventional liquid foundry binders in the mixture (Mixtures 61 - 64) severely aggravated the gas porosity problem and caused poor surface quality in the casting. Soluble amin-aldehyde thermosetting resin (a high solid urea type of water soluble binder), and a phenol-formaldehyde water dispersed binder were used. In conventional foundry practice, these are used mainly as core binders. It is probable that the gas porosity and surface quality problems were the result of incomplete removal of volatiles during firing of the molds, and that modified firing practice including higher firing temperature could be developed for these binders.

Mixtures 71, 72, and 73 were made using "Slurry" (a petroleum residual available from Union Oil Company, Los Angeles) in place of water. As in other series, the percentages of pitch were varied. These mixtures exhibited low green strength making pattern withdrawal difficult, particularly with complex molds. Mixtures 74, 75, and 76 also included Slurry, but no starch or cement. These mixtures also exhibited low green strength. Additional mixtures were made without starch, cement, or Slurry; with varying amounts of pitch and water; and with additions of coke flour.

Some of the mixtures made contained coke as a substitute for granular graphite in an effort to reduce mold cost. However, coke was unsatisfactory because of severe reaction with the molten titanium, apparently because of high sulfur in the coke.

Mixtures 1 and 17 were considered the best of those tested. They handled well during ramming, provided good casting surfaces and had desirable strength levels. Mixture 17, with less moisture than mixture 1, had superior flowability and produced the better casting surface of the two but because of its lower green and fired strength, required more careful handling of the mold. Mixture 17 was selected as the most satisfactory mold material and was used for the pilot production phase.

Shell Graphite Molds and Cores

A partially successful program was conducted to develop a shell process for making graphite molds and cores. The resulting shell was used primarily for cores, and was satisfactory except when exposed to a large volume of metal or when a sharp corner of the core was exposed to metal on both sides, in which cases penetration of metal into the shell occurred.

The experimental shell graphite mixes used in this work are listed in Table J7. The mixture selected as best and used in production of Phase IV castings is mixture 4.

The mixtures which contained calcined coke as a substitute for granular graphite produced excellent surface finishes but sometimes caused reaction with the molten titanium, apparently because of sulfur in the coke.

The investment cycle found to be best was 8 seconds, with a one minute cure. Variation in investment time directly causes variation in thickness of the shell. Curing longer than one minute does no harm but was unnecessary. After a one minute cure, the shell was quite strong and was durable enough to be handled without undue breakage. Investing and curing are accomplished at 350° F. At temperatures above 350° F, the shell adhered tightly to the pattern and was difficult to remove. Various parting agents were tried but did not eliminate the stickiness problem at high curing temperatures.

Air pressure used to force the mixture into the corebox was found to be an important factor in producing good shells. To obtain consistently strong and dense shells, it was necessary to use an air pressure of 20 - 25 psig. A problem was experienced in some trials, because of the rapid set-up of the mixture. During the build-up of the shell, a bridging of the initial layer over parts of the corebox was observed, particularly over sections that had re-entrants such that the mixture entering did not impinge directly on all portions of the corebox. Venting along the parting line of the corebox aided in reducing this problem. When molds or cores containing bridged areas were fired and used, metal penetration occurred into the bridged portions of the mold.

Shell cores were experimentally used to core the cavities in the Sway Brace Case, the Torsion Fitting, and the Flap Track Link, and to form the pockets in the Developmental Bracket. These experiments demonstrated that the shell cores are not suitable adjacent to heavy sections or to form sharp fillets, because of penetration of metal into the core. The heavier volume of metal around the shell apparently does not permit the initial chilled skin to remain in place next to the core wall. Consequently, molten metal remains in contact with the shell and penetrates into it.

Coreboxes were machined from aluminum or block graphite. The shells prepared in graphite coreboxes were equal in quality to those made in the conventional aluminum coreboxes. Graphite coreboxes were preferred for experimental work because of the ease of making dimensional changes. The graphite and aluminum boxes were initially prepared for investing by application of a silicone parting compound (grease) and then spraying with a silicon-water emulsion to facilitate shell removal. The silicon-water emulsion was re-applied before each investment cycle.

The shell graphite process was further studied to determine if complete molds could be produced in multiple stacks. A corebox was prepared to make multiple molds for casting discs. The pattern equipment consisted of two 1/2" x 9" x 10" aluminum plates separated by a 3/4" x 1" frame work around three sides to make the box. The cavity for the casting was a 5/8" x 6" diameter disc, half of the disc on the interior of each of the box halves. A central 1" diameter core was also located to provide the sprue in each of the molds. The graphite material was invested from the bottom of the box in the shell core machine. The completed mold was then fired at 1650°F in the standard manner. The shell molds, when stacked together, formed a complete 5/8" x 6" diameter disc with one-half of the disc in the cope section of each mold and one-half of the disc in the drag section; the parting line of the casting being between the two surfaces of the molds. The sprue was centrally located to permit a single gate at the bottom to feed the entire stack of castings. Twelve 5/8" x 6" diameter discs were to be cast in a single stack. Two attempts were made to pour the stack of molds. The first attempt failed because of severe mold reaction between titanium and coke molds. The second pour was partially successful but insufficient metal was poured to fill the entire set of plates and only ten of the twelve

discs filled. The second series of molds was made of granular graphite material. There was no penetration into the graphite material and the castings were relatively clean with no surface burning.

Conclusions reached as a result of the shell mold studies are as follows:

- (a) The shell molding process offers excellent potential for production use.
- (b) At the present stage of development, shell graphite molds or cores are not suitable for shapes with relatively large volume concentration or with sharp fillets, because penetration of metal into the shell occurs in these areas.
- (c) The feeding distance in shell graphite molds appears to be approximately 1.5 times that typical of rammed graphite molds.

Investment Molds

The initial experiments with investment casting techniques were conducted with a proprietary investment mold developed by Misco Precision Company, Muskegon, Michigan.

The results of the first investment casting trial are shown in Figure B7. The castings had good reproduction of detail, but rough surface and some surface burn-in. The major defect was the severe gas porosity in the casting. Figure B8 is a print of the x-ray of the tensile test specimens and clearly shows the extreme porosity.

Figure B9 is a photograph of a casting made in another type of investment mold which was made with a graphite primary coat. Gas porosity was extensive throughout the castings, but surface finish was excellent. The mold was centrifugally cast in the vertical axis centrifugal furnace in an attempt to eliminate the gas porosity problem. The ingates into each of the molds broke and complete mold filling did not occur. The threaded portions of the test bars showed evidence of reaction between the mold material and the metal.

Castings made in a third type of Misco Proprietary investment mold showed severe surface reaction and gas porosity. The primary surface of this mold was composed of graphite and zirconium oxide. The type of binder used is not known. The photograph in Figure B10 illustrates the severity of reaction.

A general observation in connection with experimental casting of these molds is: to obtain a gas free casting, a binder that does not retain oxide materials must be used to prevent reaction and subsequent gas in the castings.

Additional investment casting trials were conducted using molds developed by Oregon Metallurgical Corporation. The program was aimed at producing an experimental mix that would not react with the metal and that would produce gas-free castings. The mixtures tried are listed in Table J8. Casting trials were conducted on only those mixtures that showed promise as an investment material.

Castings were poured into simple bar molds made from mixtures 11, 14, 15, and 16.

The molds made from mixtures 11 and 15 exhibited violent reaction with the molten titanium, only about half the metal remaining in the mold. It is believed that moisture in the second coat caused the reaction. The mold made from mixture 14 also reacted violently, with no metal remaining in the mold. This is believed due to the oxides and silicates in the second coat.

Mold mixture 16 was used to make four trial castings with no evidence of reaction. A problem of mold-face porosity was caused by bubbles in the mixture and resulted in rough surface on the casting. It is expected that further development of the process (such as mixing of the ingredients in a vacuum chamber and mechanical vibration during investment) would solve this problem.

In general, the development of a satisfactory investment mold for casting of titanium appears feasible if based on the use of graphite.

Recycling of Mold Materials

The object of this study was to determine if mold materials could satisfactorily be reused to make new molds and, if so, to establish suitable preparation methods to obtain proper grain size and distribution in the recycle materials.

The process for reclaiming the rammed graphite molds involves crushing, pulverizing, and screening. The crushing operation was to reduce the size of mold fragments broken from the castings to approximately 1/2-inch size for feeding to the pulverizer (a hammer mill) which then reduced the material to the proper grain size for making new molds. In some instances when crushing machined graphite, a large proportion of fines were produced and screen classification was necessary to separate the useful portion from the fines.

To establish a procedure for obtaining material of proper grain size from the pulverizer, several samples were processed using different rotor speeds and screens, these being the variables most effective in adjusting grain size. Table J9 is a tabulation of grain size distributions of the test samples. The rotor speed of the pulverizer was varied between 14000 and 6070 RPM. The higher rotor speeds produce larger proportions of fines. The rotor speed of 6070 RPM combined with the 1/8-inch mesh herringbone screen produced the lowest percentage of fines.

The present stage of reclamation of the pulverized graphite mold material involves a screen classification process to obtain the desirable portion of the pulverized material. The material as recovered from the pulverizer can be used in two different ways. The first is to use that material which will pass through the No. 20 screen of a vibratory shaking unit. The distribution of the material is listed as Sweco No. 20 in Table J4. This material was somewhat coarse, but produced acceptable molds where fine surface finish was not essential. The most satisfactory combination of classified material has been found to be 60 percent Sweco No. 20 and 40 percent Sweco No. 40 mixed together and used with the standard mix ingredients. This combination contains a higher portion of fines than is normally desired in general foundry molding practice, but has been found to be satisfactory for the rammed graphite mold practice.

FIGURE B6

CARBON CONTAMINATION DEPTH IN TI-6Al-4V CASTINGS

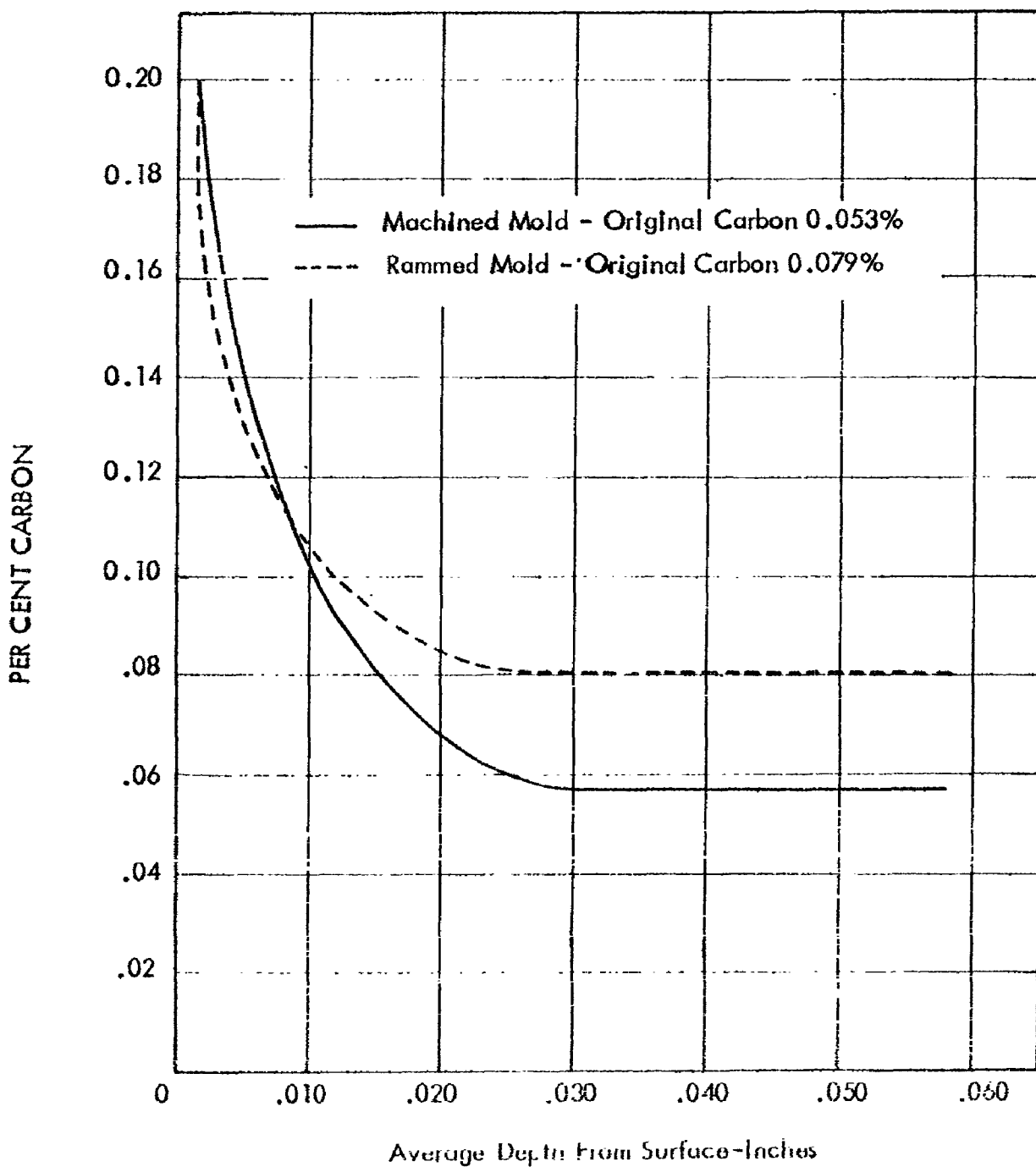


FIGURE B 7
INVESTMENT CASTING

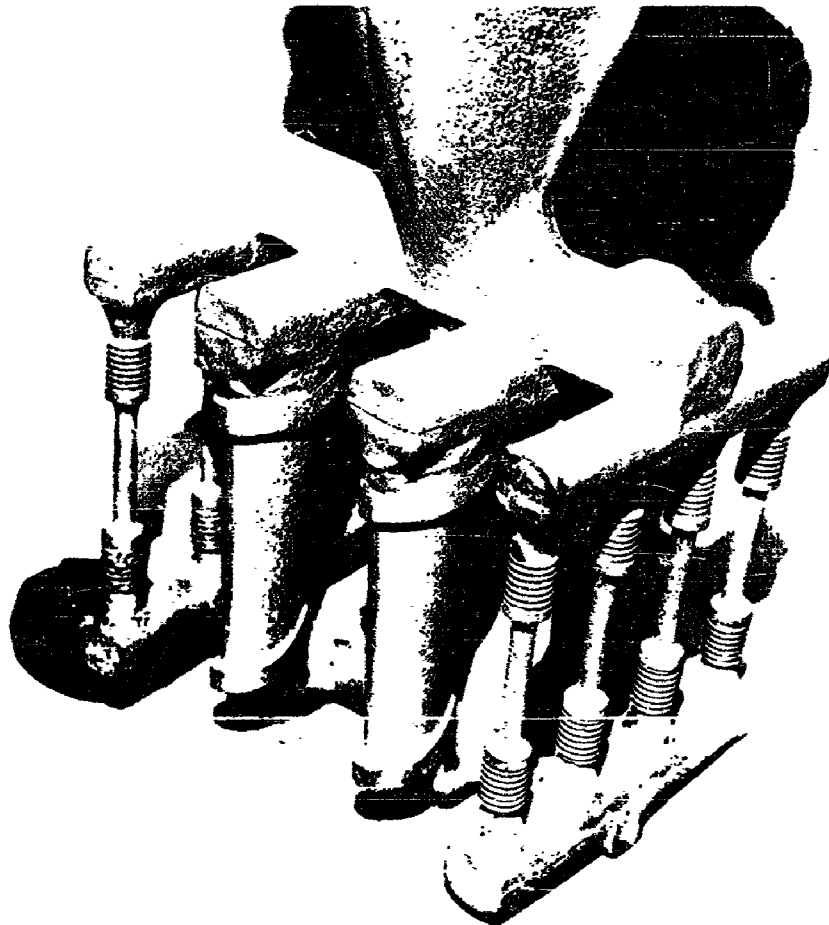


FIGURE B8

X-RAY OF INVESTMENT CASTING

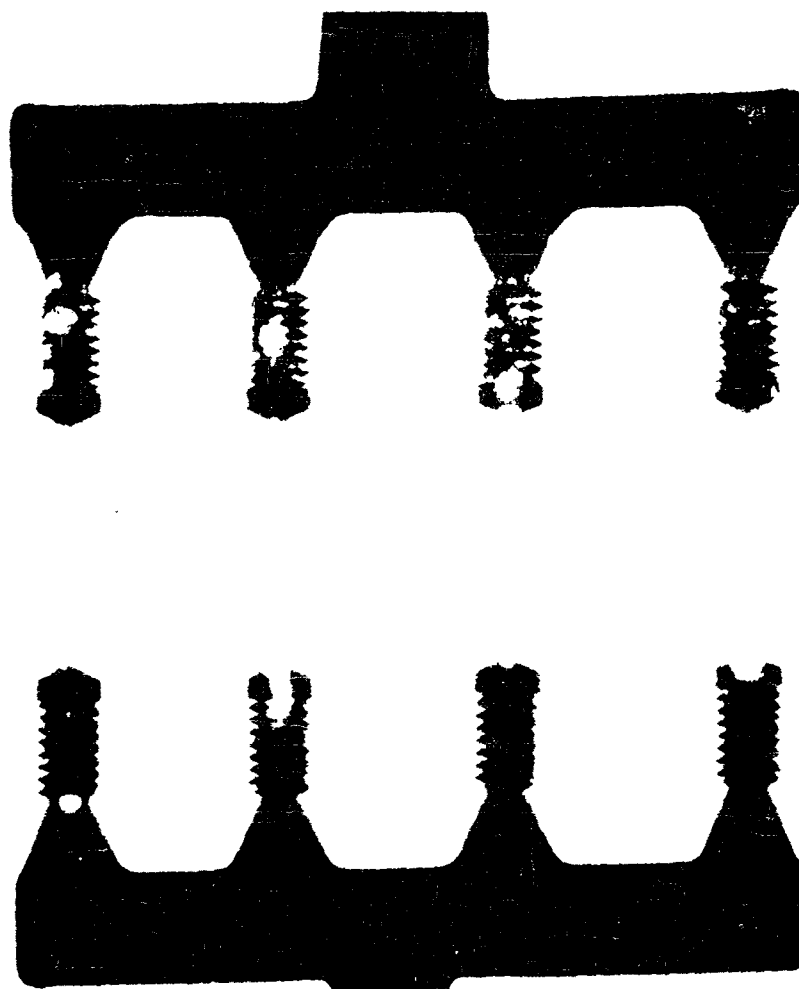


FIGURE B9

INVESTMENT CAST Ti-6Al-4V



FIGURE B10
INVESTMENT CAST Ti-6Al-4V



FEEDING STUDIES

Feeding Distance In Flat Sections Cast In Machined-Graphite Molds

An early study in this program consisted of the casting of Ti-6Al-4V titanium alloy into a series of round, flat discs of various diameters, thicknesses, and riser sizes, using machined graphite molds. The purpose of the study was to become familiar with the feeding characteristics of titanium alloy castings.

The specific objectives of the study were:

- (a) to determine minimum riser diameters required to prevent under-riser shrinkage porosity,
- (b) to determine the "edge effect", or soundness contributed by faster cooling adjacent to the disc edge,
- (c) to determine feeding distance changes caused by risering variations,
- (d) to measure the total effective feeding distance contributed by the riser and by edge effect.

Plate diameters of five and six inches were cast, with thicknesses from 3/8 to one inch in 1/8-inch increments. Riser diameters were equal to their heights and were varied from one to 2 1/2 inches. In all cases, the risers were concentric to the plates. A typical casting heat is shown in Figure B11. One inch gates were used. All molds for this series were machined from graphite block. A typical mold setup is shown in Figure B12.

After casting, the plates were x-rayed and examined for under-riser shrinkage, shrinkage beyond the riser, and soundness adjacent to the edge and riser. The total soundness is expressed as "T" and is equal to D/t where D is the sum of the sound distance measured radially from the edge of the riser and from the edge of the plate, and t is the thickness of the plate.

The results of the study are tabulated in Table J10.

After the first three trial pours, it was established that bottom gating, as shown in Figure B11, produced the best casting detail. This general setup was used for making all subsequent static pours of this type.

Examples of extreme conditions of shrinkage observed are illustrated in Figures B12, B13, and B14. These illustrations are photographic prints made directly from x-ray films.

The following was concluded from this study:

- (a) Feeding distance in flat sections (no taper) in unheated machined-graphite molds does not exceed 1T for consistently sound castings.
- (b) Riser diameter at its junction with the casting must be not less than two times the casting thickness, to prevent under-riser shrinkage in castings made in unheated machined-graphite molds.
- (c) Bottom gating provides casting detail superior to top gating.

Feeding Distance In Tapered Sections Cast In Machined and In Rammed Graphite Molds

A second feeding distance study was conducted to evaluate the effect of taper on soundness of cast titanium discs. In this study, both rammed-graphite and machined-graphite molds were used. All discs were tapered such that thickness increased from edge to center, and were cast in edge thicknesses of 1/8 to one-inch in 1/8-inch increments. The following types of static molds were used:

- (a) unheated machined-graphite,
- (b) unheated rammed-graphite,
- (c) heated machined-graphite,
- (d) heated rammed-graphite.

The results obtained on study of x-rays of the castings made in unheated machined-graphite and rammed-graphite molds are presented in Tables J11 and J12. The parameter "T" (defined on page B30) is plotted against taper for various disc edge thicknesses in Figures B15 through B17. Figure B18 relates the taper required for sound castings to the edge thickness of the discs, for castings made in machined or in rammed-graphite molds.

The results obtained in similar trials using heated machined-graphite molds are in Table J13. These tests were made using discs of 1/8 to one inch edge thicknesses with zero, one, three, five, and seven-degree tapers. The mold temperature at the time of pouring varied from 210 to 355 degrees fahrenheit. The parameter "T" for the heated molds is plotted against taper for various edge thicknesses in Figures B19 through B21, along with the "T" values obtained from the similar tests of unheated machined-graphite molds.

A similar series of trials was conducted using heated rammed-graphite molds. The mold temperature at the time of pouring varied from 200 to 400 degrees fahrenheit. The results obtained are tabulated in Table J14 and plotted in Figure B22 to B24.

The following observations were made from this study:

- (a) Increased taper in unheated machined-graphite molds improves soundness primarily by increasing the edge effect. Increased taper in unheated rammed-graphite molds improves soundness primarily by increasing riser effect. Less taper is generally required in rammed-graphite molds than in machined-graphite molds, for equivalent soundness.
- (b) Shrinkage porosity in cast titanium appears as distinct voids rather than as cloudy low-density areas on the x-ray film. Microshrinkage has not been observed.
- (c) In cast tapered discs of 1/8-inch edge thickness, riser feeding distances were greater in machined-graphite molds, however, for all other thicknesses tested the rammed-graphite molds provided the greater riser feeding distance.

- (d) Increasing the amount of taper progressively improves feeding distances in rammed molds. In the case of machined-graphite molds the feeding distance decreases until taper exceeds three or four degrees, and then increases.
- (e) The effectiveness of taper decreases as the section thickness is increased.
- (f) The tapers required to cast sound sections have been established in relation to thickness, and are shown in Figure B18 in graphical form for machined and rammed-graphite molds.
- (g) The use of heated molds did not appreciably improve the feeding of cast Ti-6Al-4V alloy but did decrease the gas porosity problem, apparently because of reduced moisture pickup during mold assembly.

Feeding Distance in Shell Graphite Molds

The experimental castings of plates in shell graphite molds exhibit greater areas of soundness than castings in the other mold materials. The initial casting of the plates was primarily for determination of carbon contamination from the shell graphite material. Due to excessive distortion of the shells during casting, the plates were not used for carbon contamination studies. X-rays of these plates revealed only small dispersed shrinkage areas, as contrasted to general heavily dispersed shrinkage in previous castings produced in rammed graphite and machined graphite in the 5/8" plate thicknesses without taper. There was evidence of gassing occurring in two of the four plates cast in the heat. The second experimental casting of shell graphite plates was made with a supporting arrangement to prevent sagging about the periphery of the plate. The casting experiment was not successful due to inadequate support on the surface of the cope, resulting in mold rupture. The chilling effect was sufficient to prevent loss of metal through the cracks in the molds even though the plates were much thicker than desired, because of the mold deformation. It appears that slight tapers in plates cast in shell graphite molds would greatly extend feeding distance, because the absence of an effective amount of chilling permits the metal to solidify in a slower and more directional manner toward the casting risers.

Feeding Distance In Centrifuge-Casting

A comparative study of feeding distance in centrifuge-cast discs was conducted using 1/2-inch thickness by six inch diameter disc molds mounted on a central sprue. Four plates were cast in machined-graphite molds in each of six heats. The casting setup is shown in Figure B25.

The object of the study was to determine the effect of in-gate size and centrifuge speed on the feeding distance. The data obtained are in Table J15.

The feeding characteristics in these discs were not significantly different from the similar discs which were statically cast, except the edge effect on soundness was somewhat less in the centrifuge-cast plates. There was not an appreciable difference in soundness when rotation speed was increased.

It was noted that the portion of disc which was forward of the gate during centrifuging had improved soundness over the trailing section. A second group of centrifuge-cast discs were made with the gates at the trailing edge. This modification did not improve casting soundness.

Several casting trials were made to investigate gating techniques in centrifuge casting. Molds were prepared to produce 1/4 by three by five inch plates, and were arranged so that the plate mold could be filled either from the leading or trailing edge, by reversing the direction of rotation of the centrifugal casting apparatus. A photograph of a typical casting produced during these trials is shown as Figure B26. When the centrifuge was rotated counterclockwise the plate molds were effectively "top filled", and were "bottom filled" when rotated clockwise. A comparison of surface qualities of castings produced at various rotation speeds (200, 700, and 1600 rpm) and directions demonstrated considerably superior surface quality in the "bottom filled" plates at all centrifuge speeds. Comparison of x-ray quality showed little difference in clockwise or counterclockwise rotation at low rotation speed (200 rpm) but considerable advantage in "bottom filling" at the higher rotation speeds.

The centrifuge casting of up to twelve 1/2-inch thickness by five inch diameter discs in a single pour was investigated to determine the practicability of the proposed multiple centrifuge casting technique. A machined-graphite mold setup was prepared as shown in Figure B27. Figure B28 is a photograph of a typical trial heat, showing the arrangement of molds, gating, and central sprue. The gates were one inch in diameter from the sprue to the two inch diameter ingates feeding the casting.

Examination of the X-rays of these plates revealed that the gating size was not sufficient to provide soundness in the discs. The locations and sizes of the shrinkage voids in the twelve discs were very consistent. Additional trials were made using three-cavity mold setups such as shown in Figure B29. Larger and smaller diameter gate and ingate diameters were tried with some improvement in soundness but complete absence of shrinkage porosity was not obtained.

There was no evidence of turbulent metal flow on the casting surface, except for a very slight surface ripple near the ingate. As the radius at this junction was increased, the surface ripple appearance decreased. The entire set of castings had consistently good surface finish.

FIGURE B11
TYPICAL FEEDING STUDY ARRANGEMENT

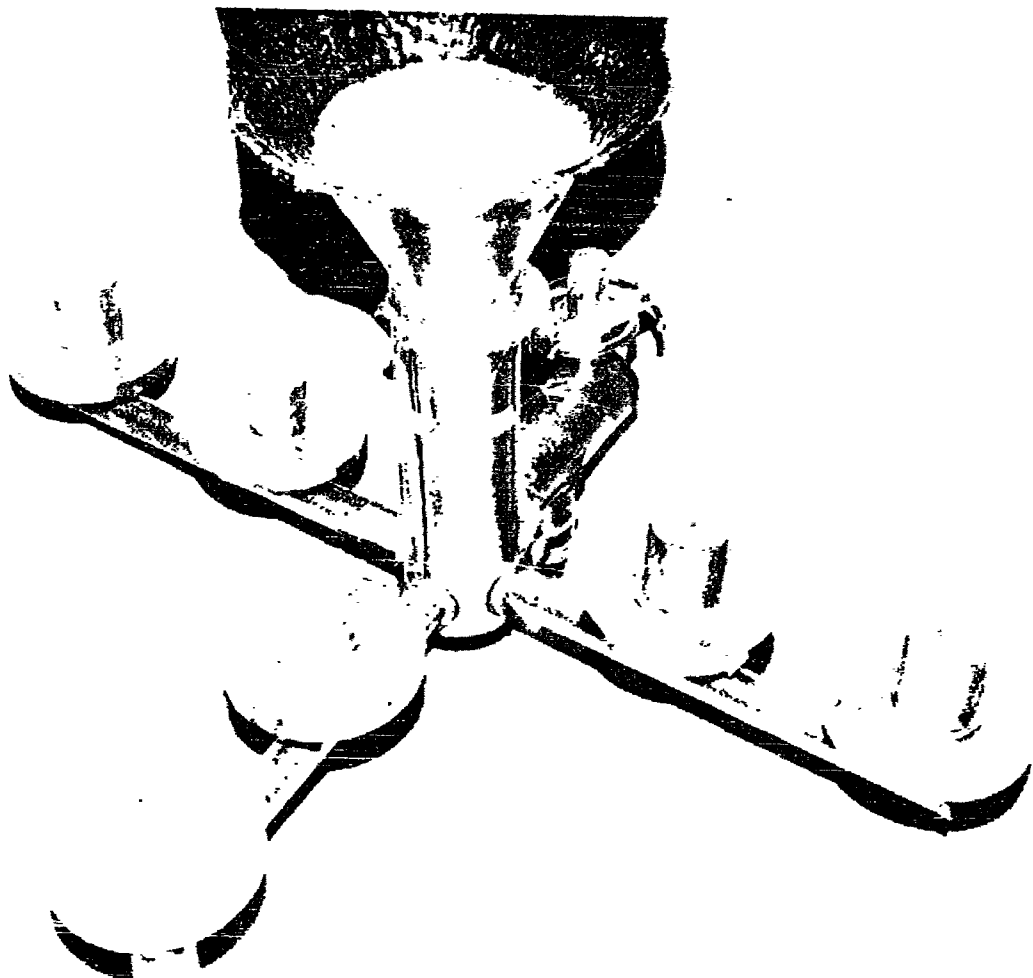


FIGURE B12a

GENERAL DISPERSED SHRINKAGE - FEEDING
DISTANCE EXCEEDED GREATLY

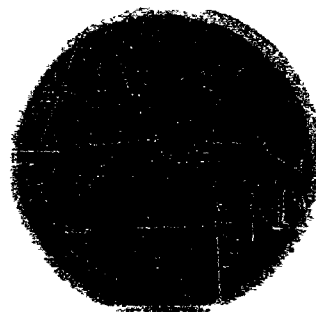


FIGURE B12

TYPICAL PLATE MOLD SETUP

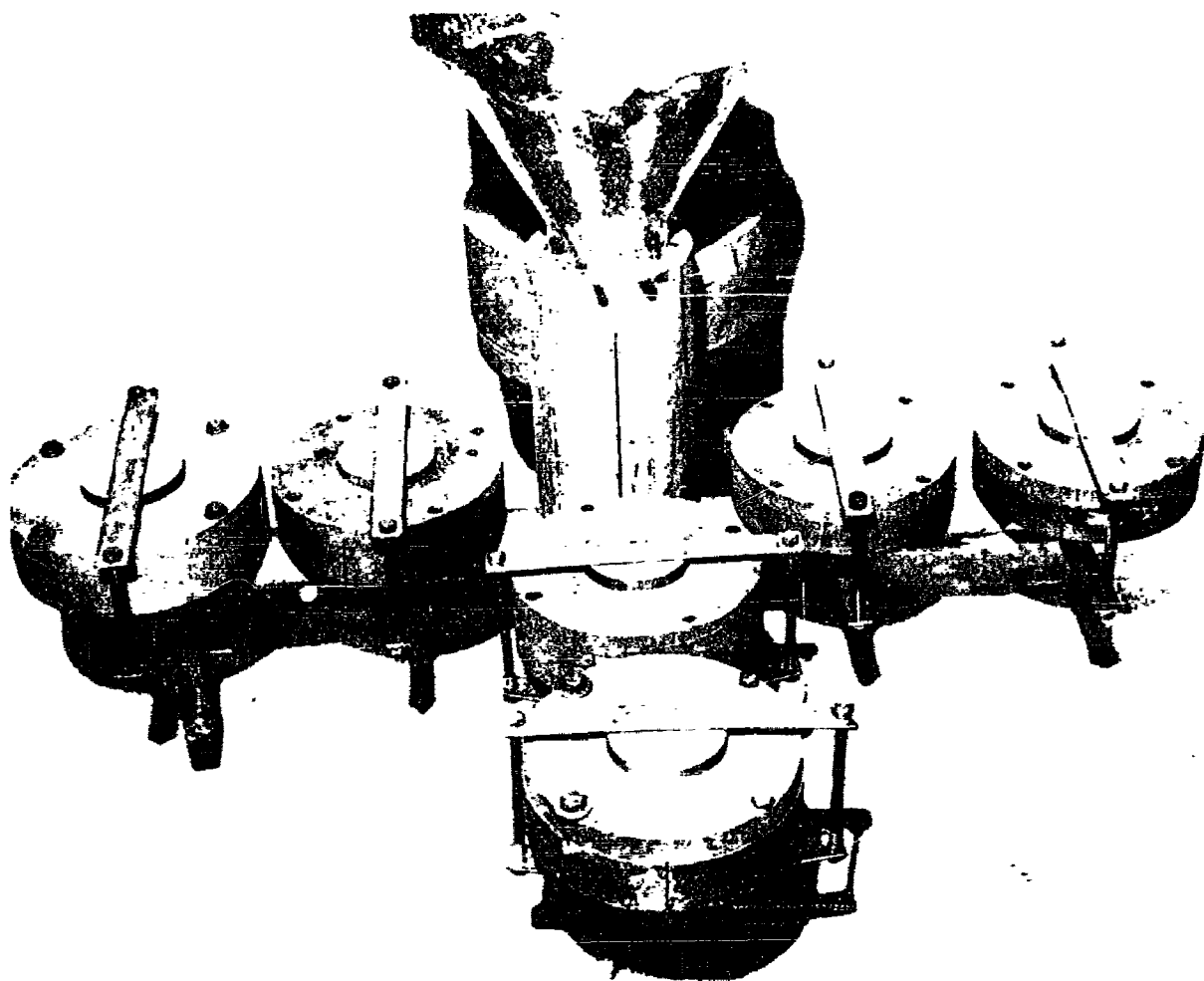


FIGURE B13

SHRINKAGE IN SLIGHTLY EXCEEDED
FEEDING DISTANCE

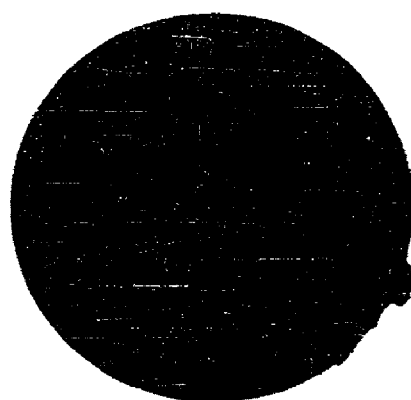


FIGURE B14

FEEDING REQUIREMENT SATISFIED

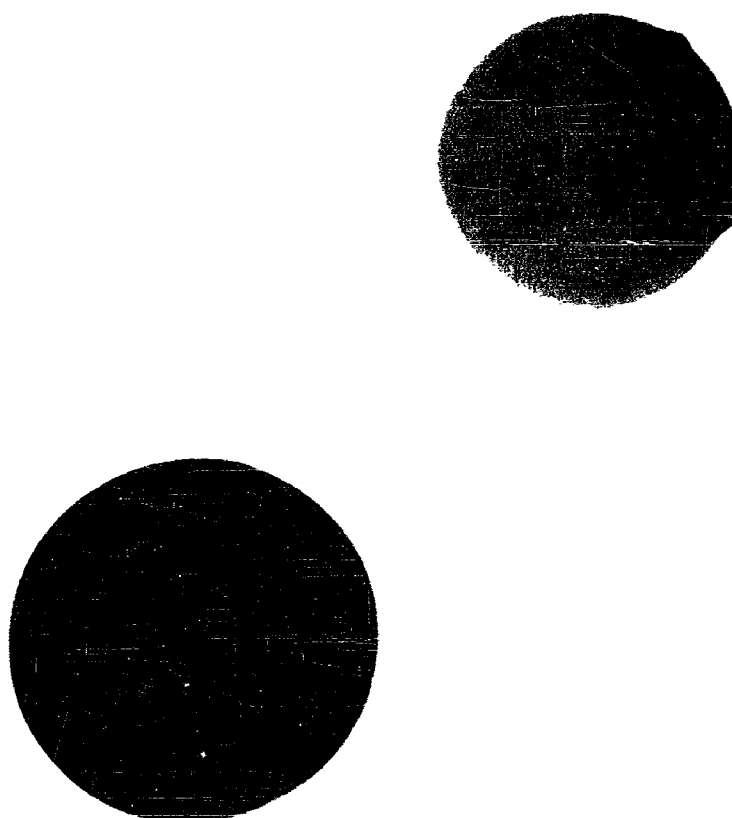


FIGURE B15

FEEDING DISTANCE IN 6" DIAMETER DISCS

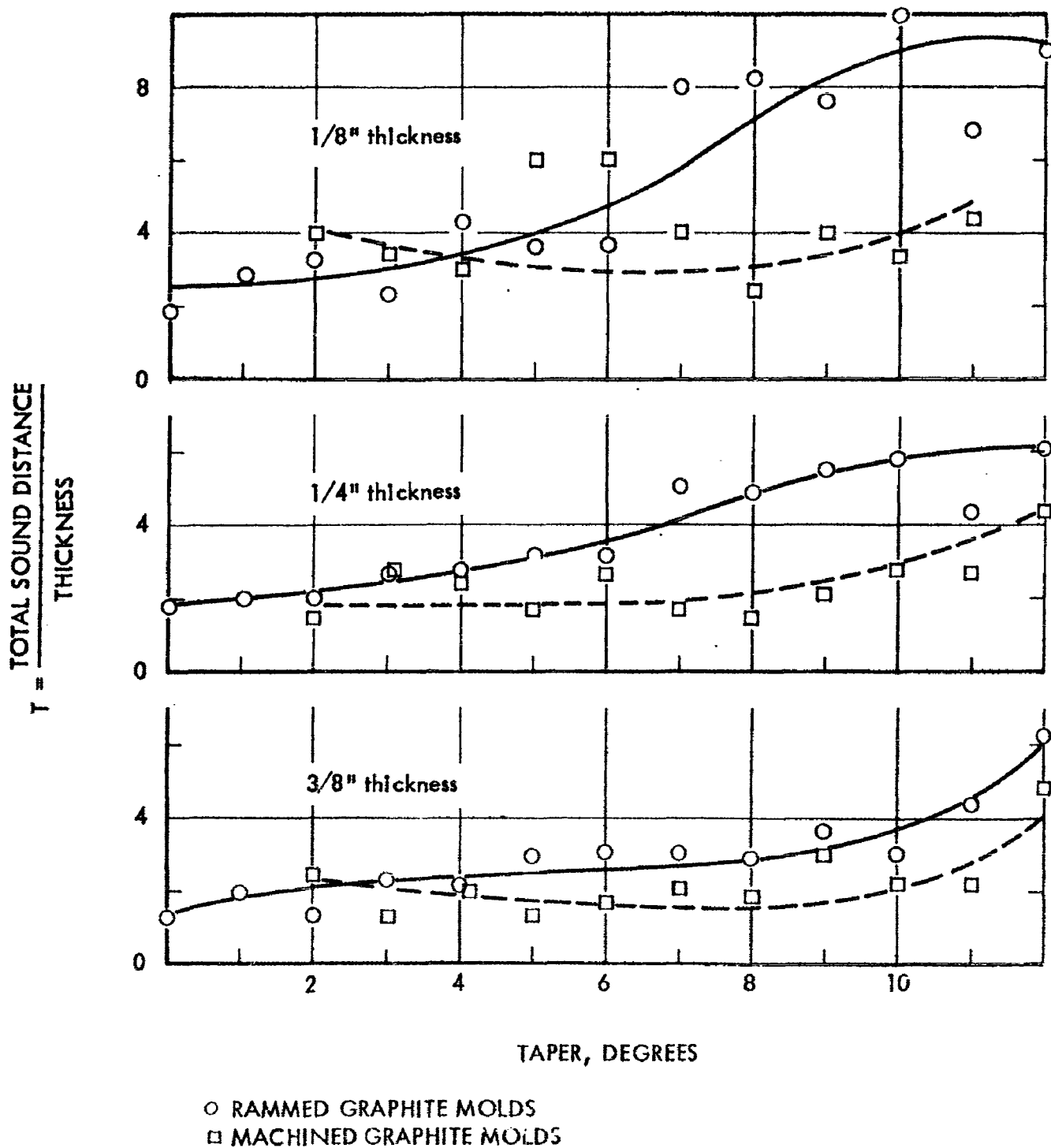


FIGURE B16

FEEDING DISTANCE IN 6" DIAMETER DISCS

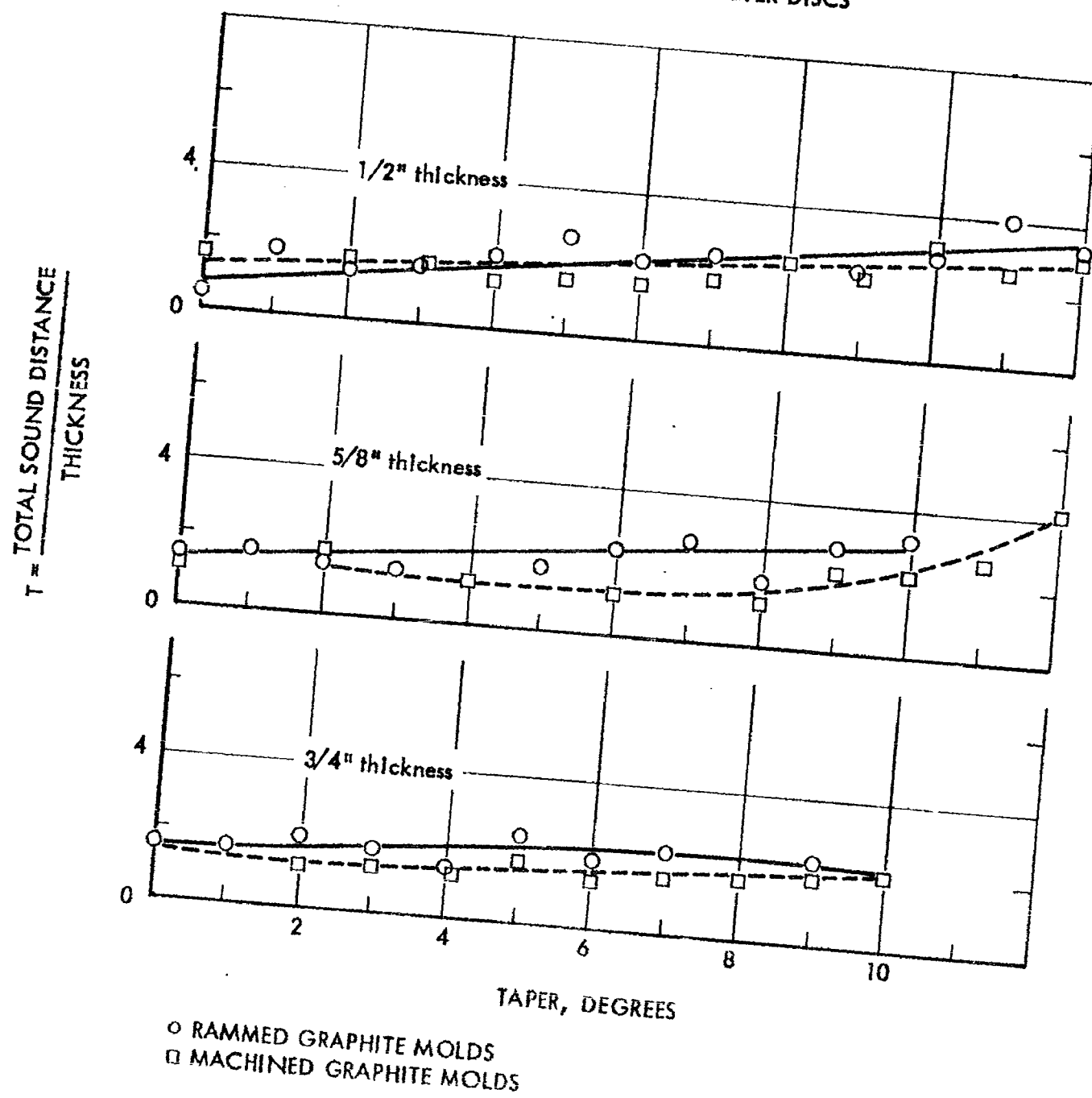


FIGURE B17

FEEDING DISTANCE IN 6" DIAMETER DISCS

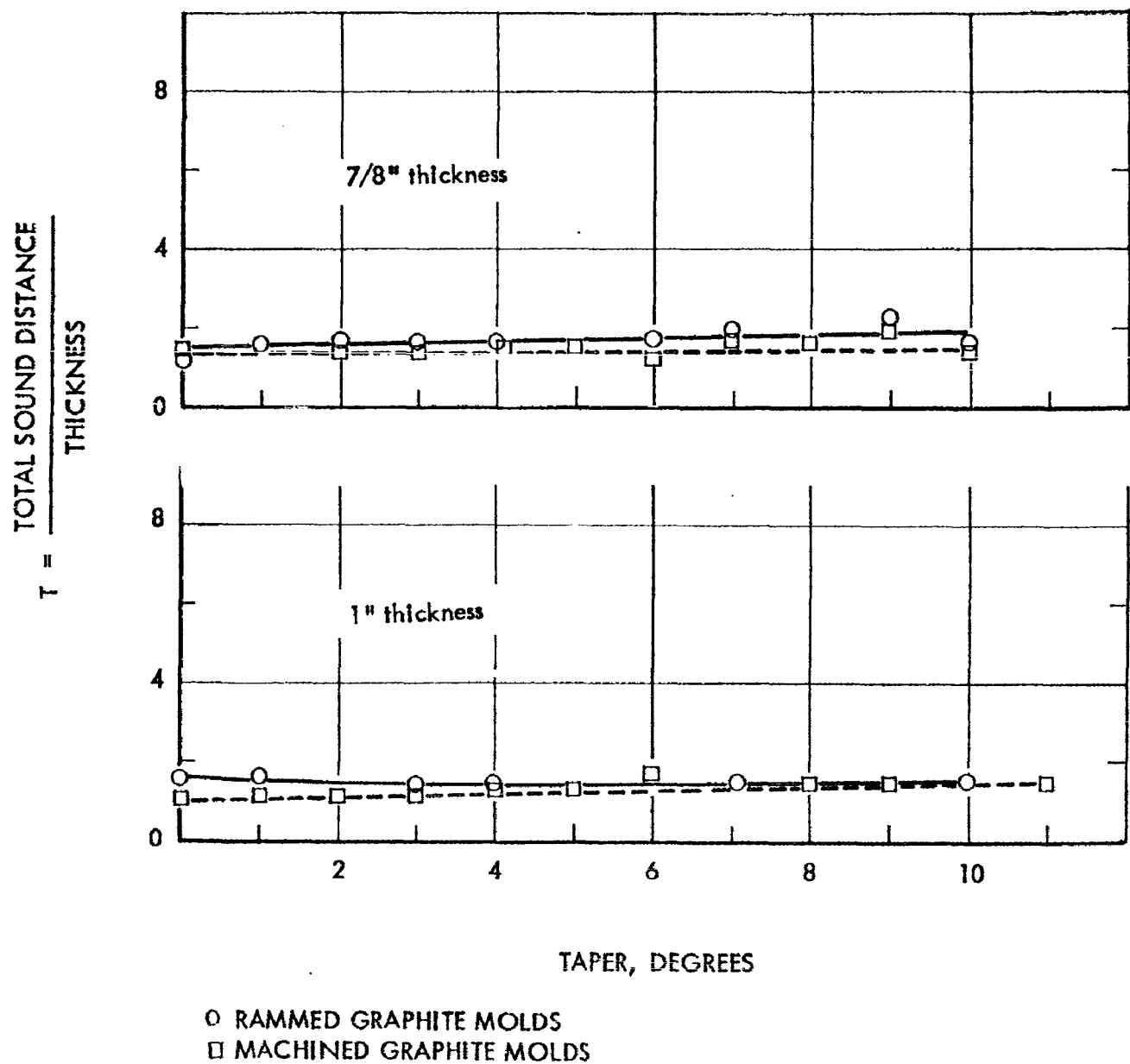


FIGURE B18

TAPERS REQUIRED TO CAST SOUND 6" DIAMETER DISCS

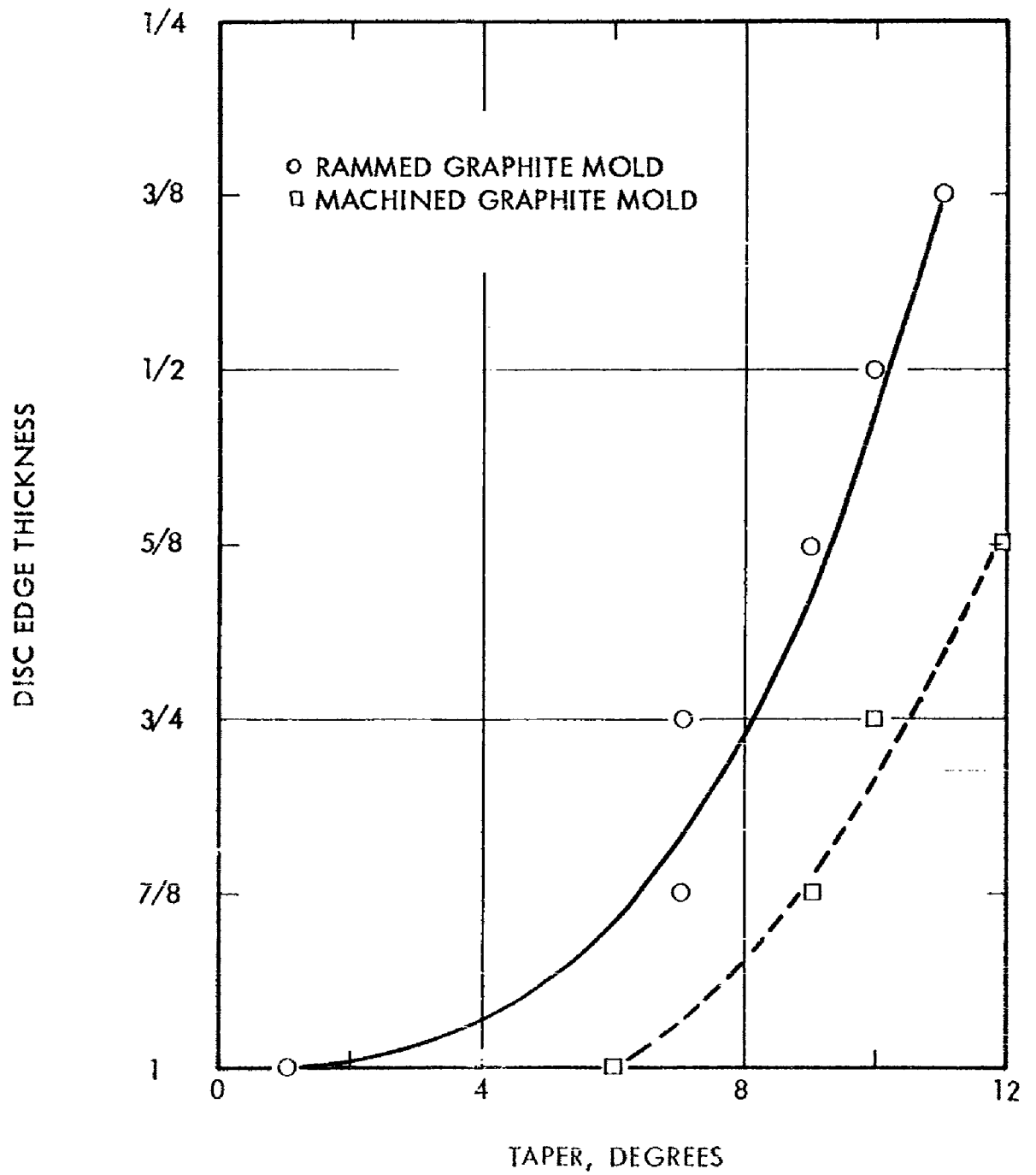


FIGURE B19

FEEDING DISTANCE IN 6" DIAMETER DISCS

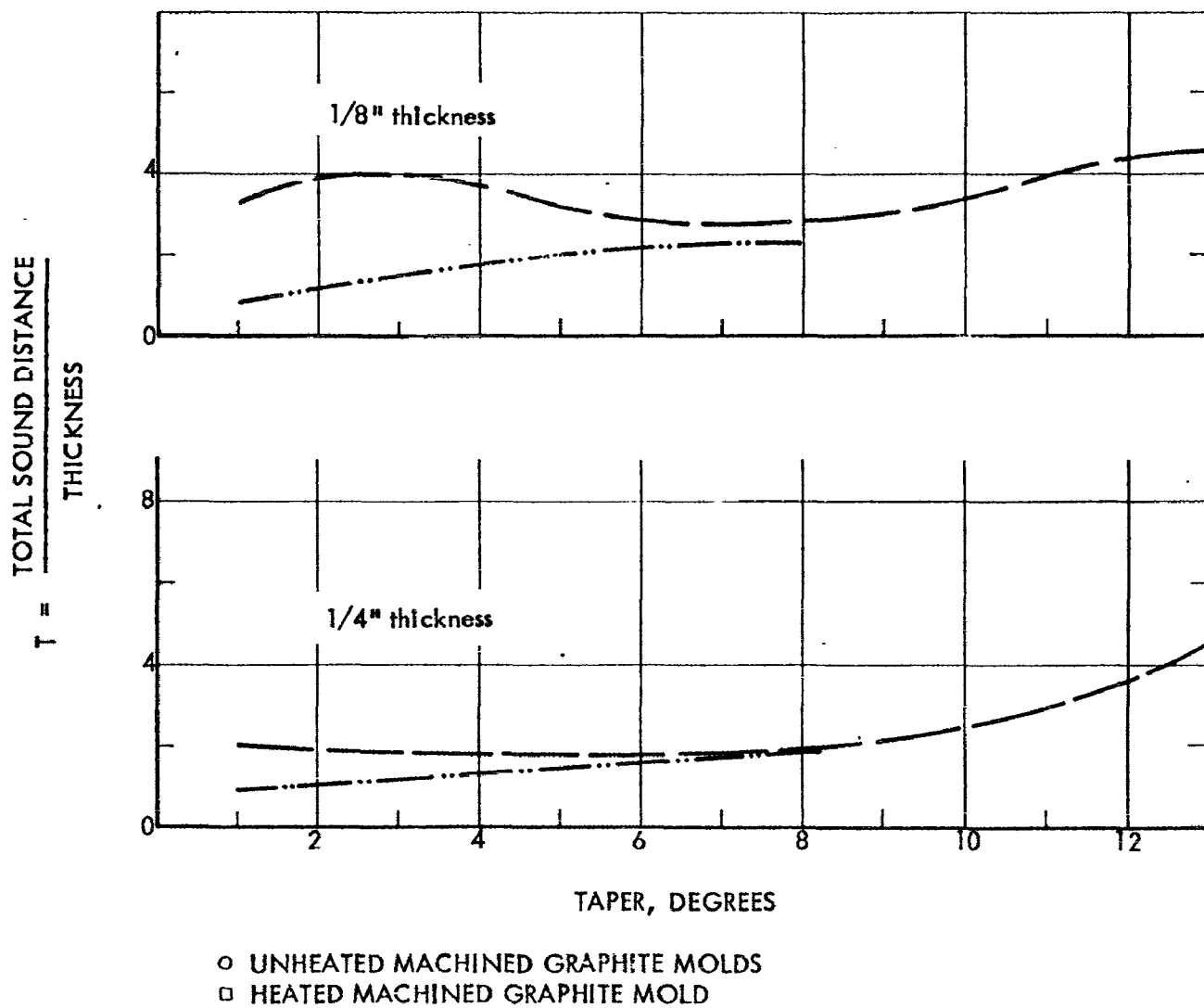


FIGURE B20

FEEDING DISTANCE IN 6" DIAMETER DISCS

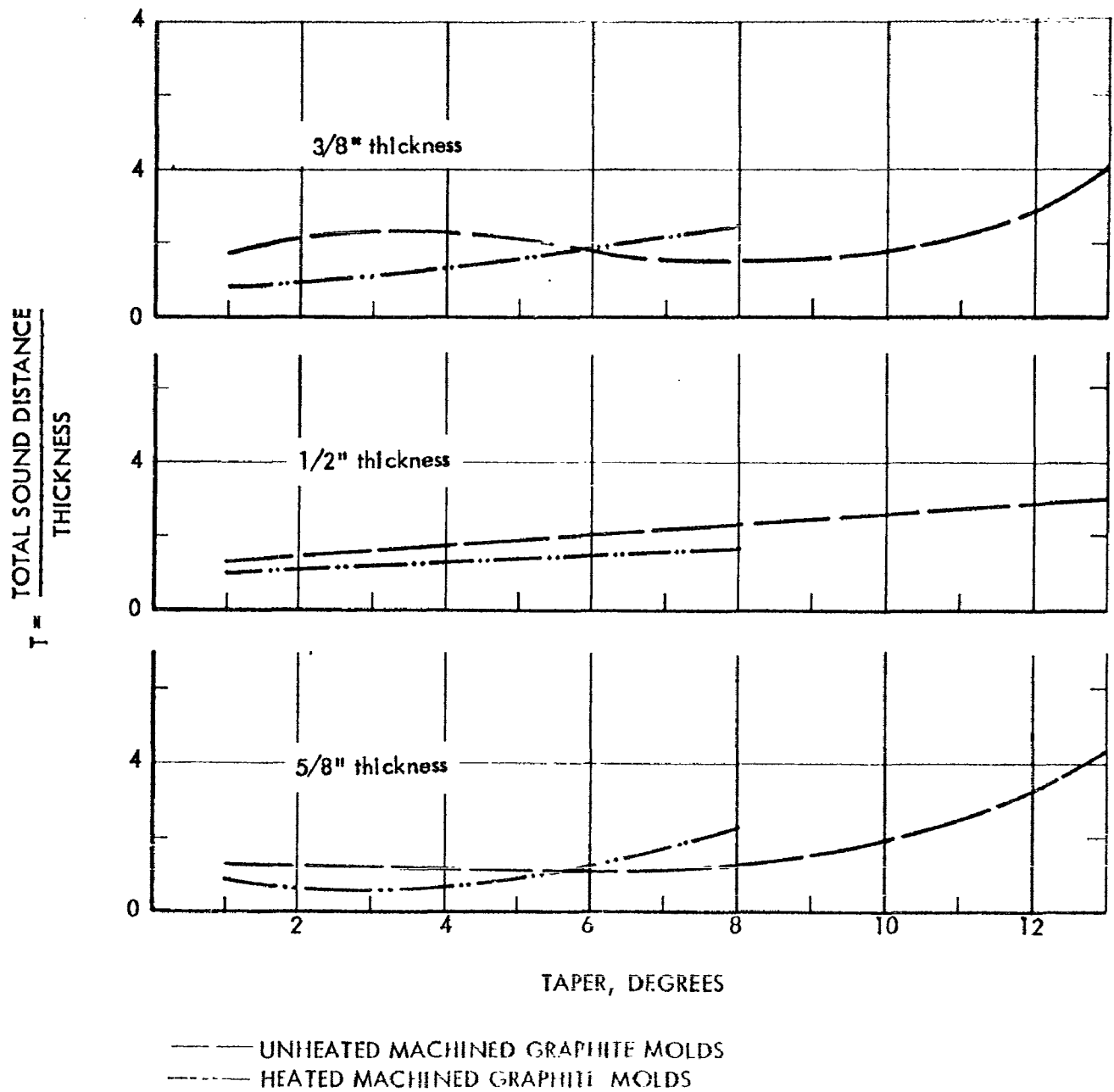


FIGURE B21

FEEDING DISTANCE IN 6" DIAMETER DISCS

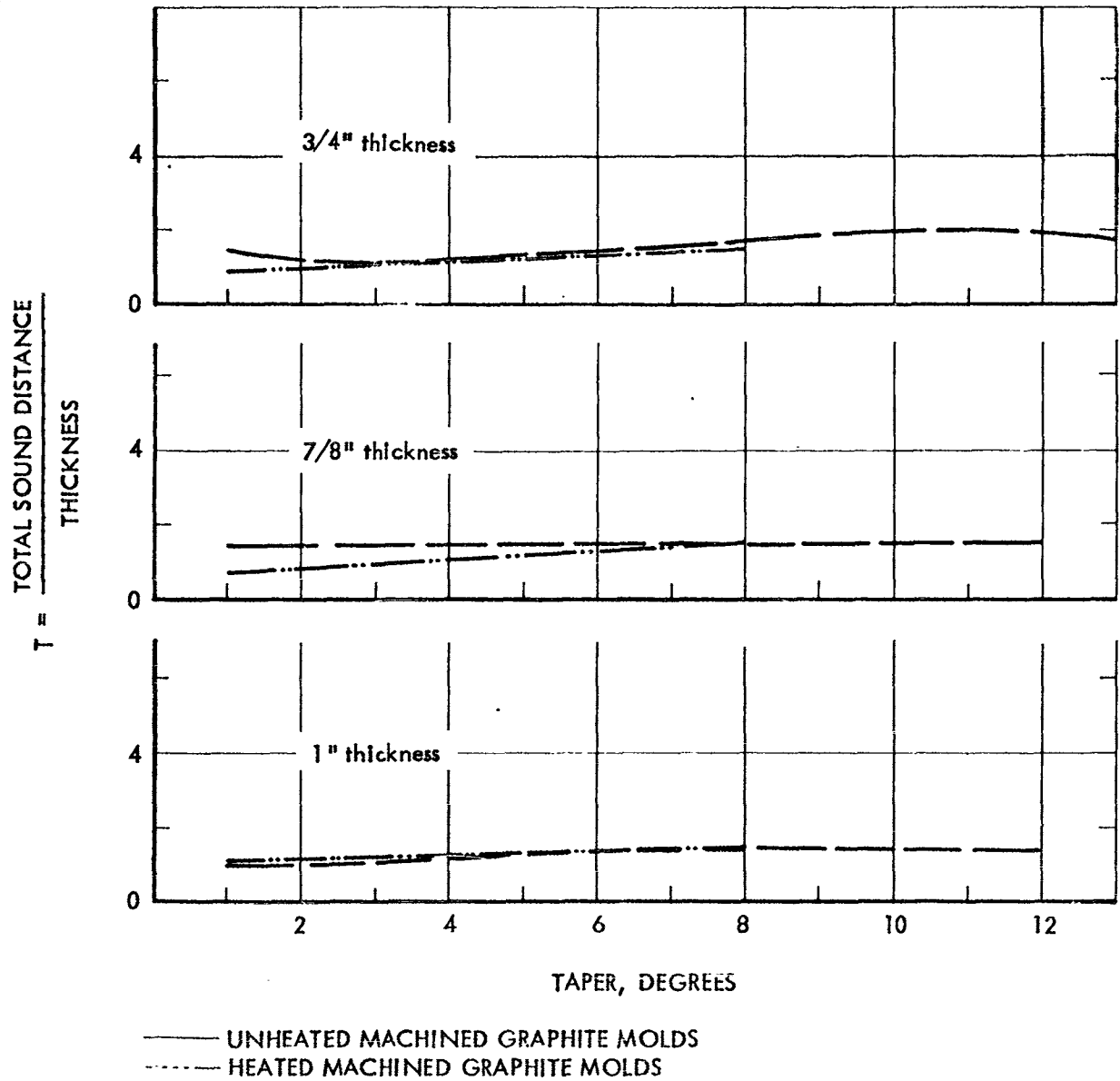


FIGURE B22

FEEDING DISTANCE IN 6" DIAMETER DISCS

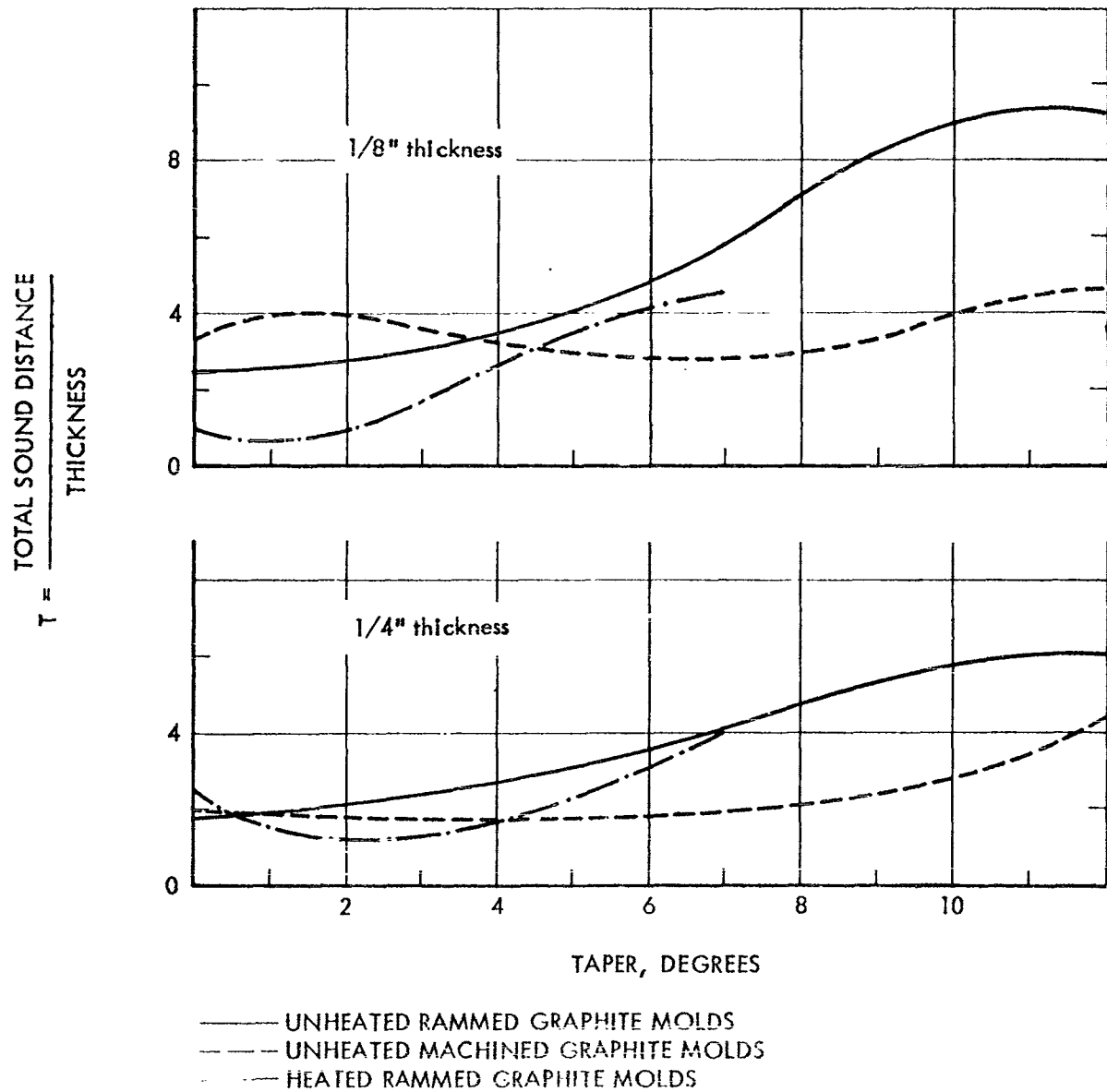


FIGURE B 23

FEEDING DISTANCE IN 6" DIAMETER DISCS

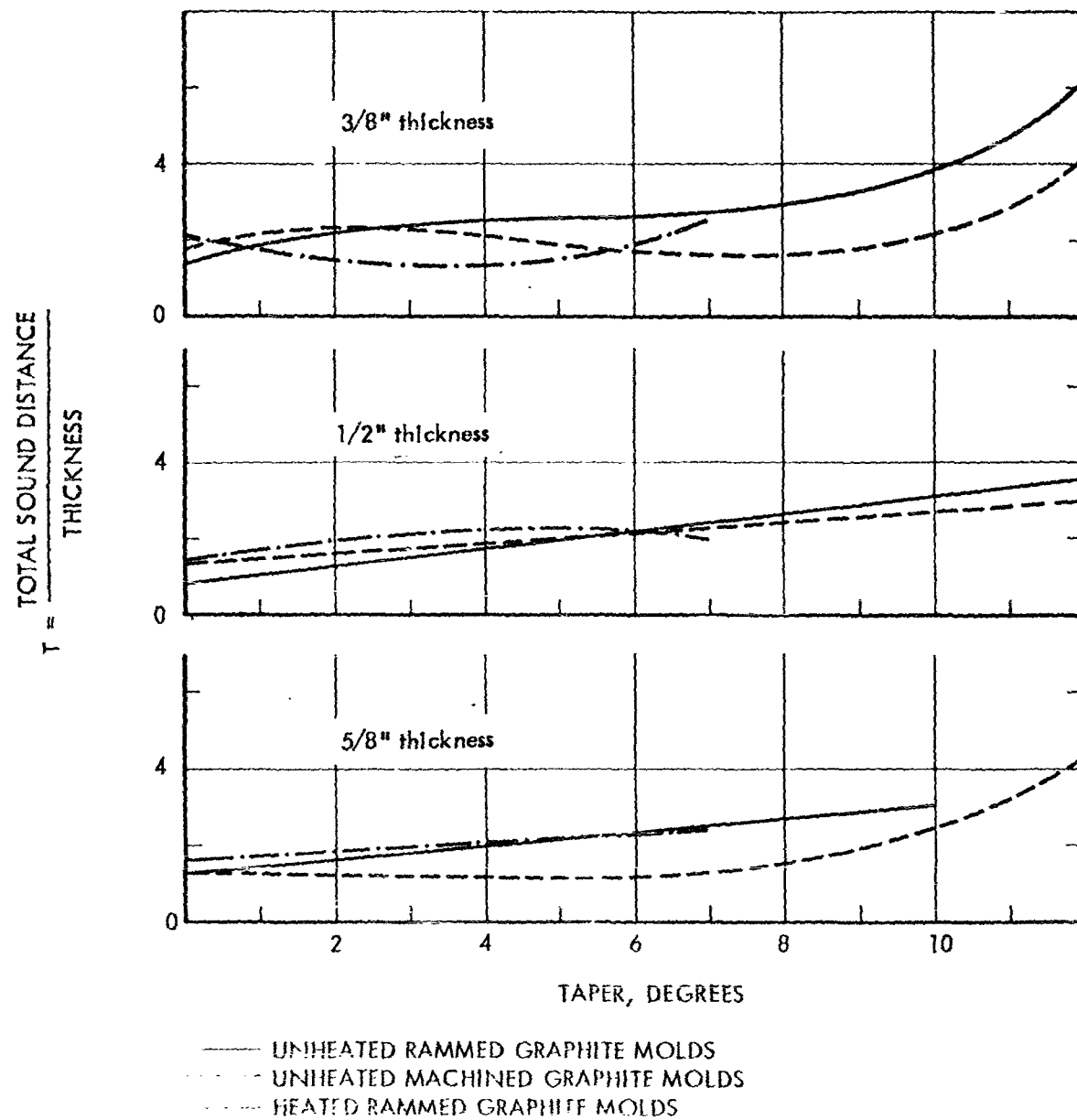


FIGURE B24

FEEDING DISTANCE IN 6" DIAMETER DISCS

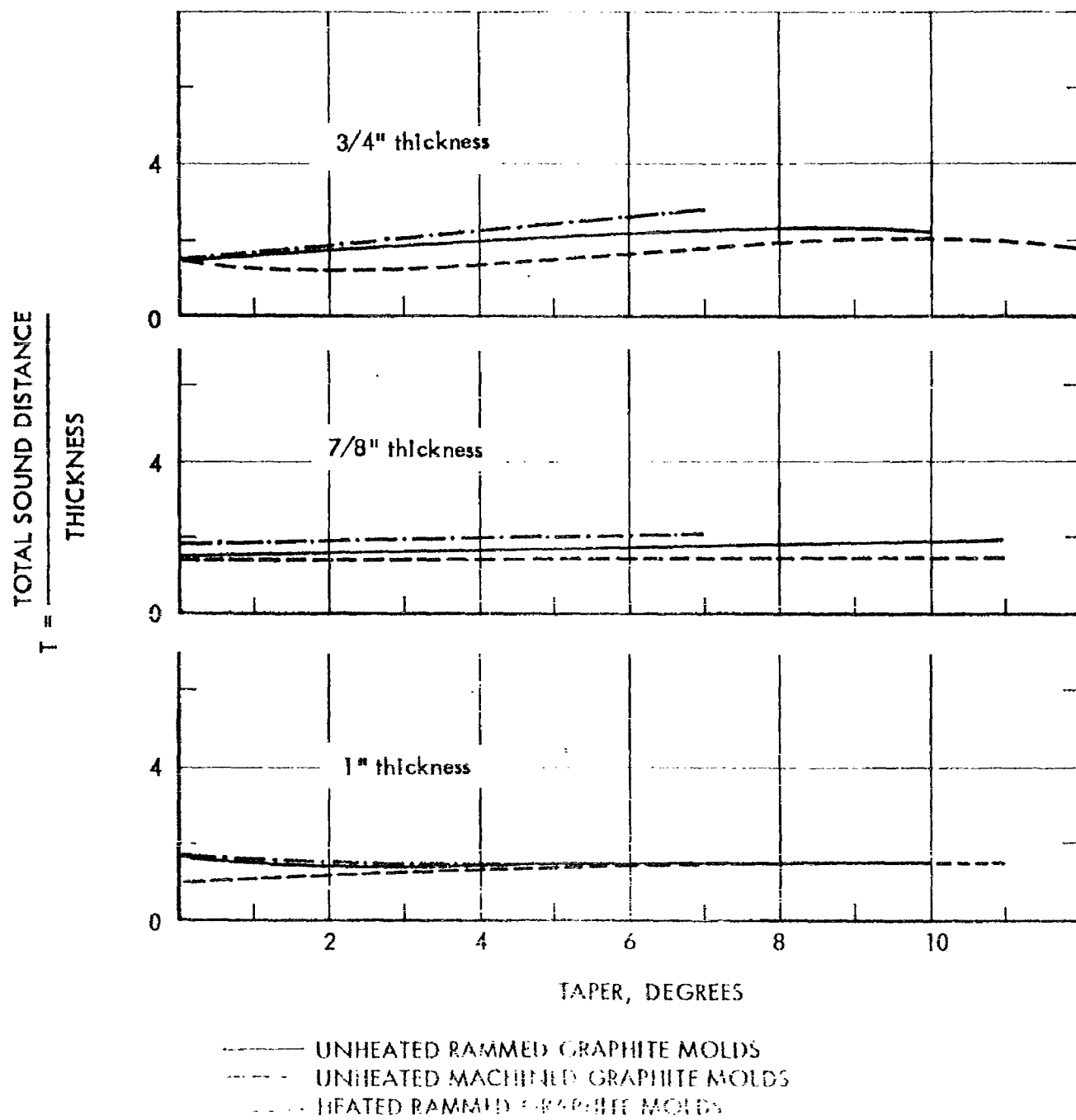


FIGURE B 25

TYPICAL SETUP FOR CENTRIFUGED PLATE CASTING TRIALS



FIGURE B26

CENTRIFUGALLY CAST PLATES

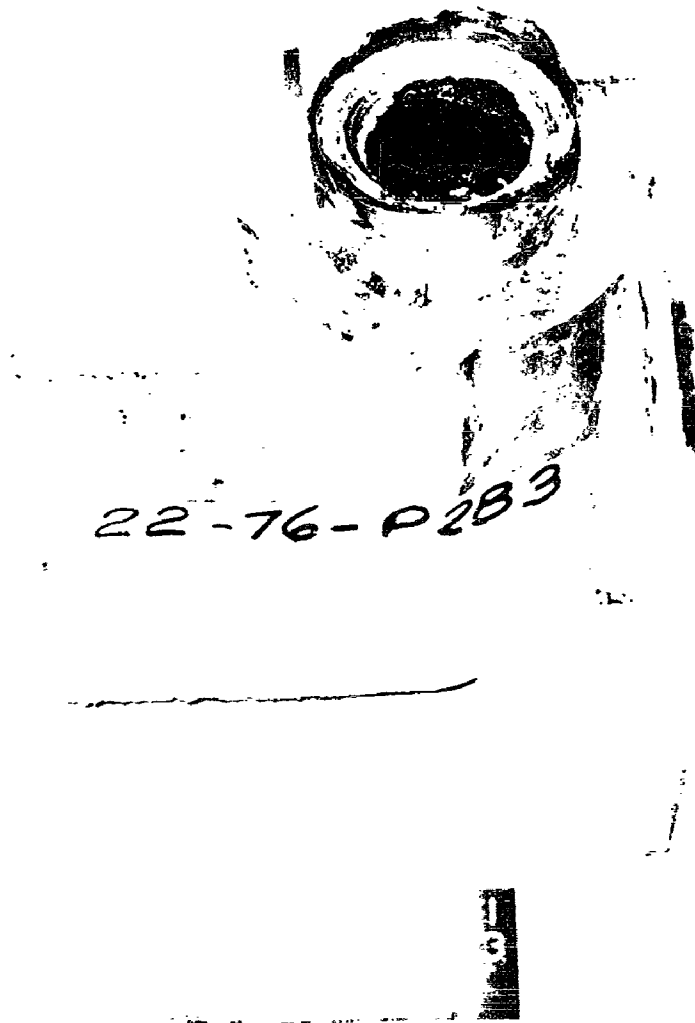


FIGURE B27

TYPICAL MOLD SETUP FOR CENTRIFUGALLY CASTING DISCS

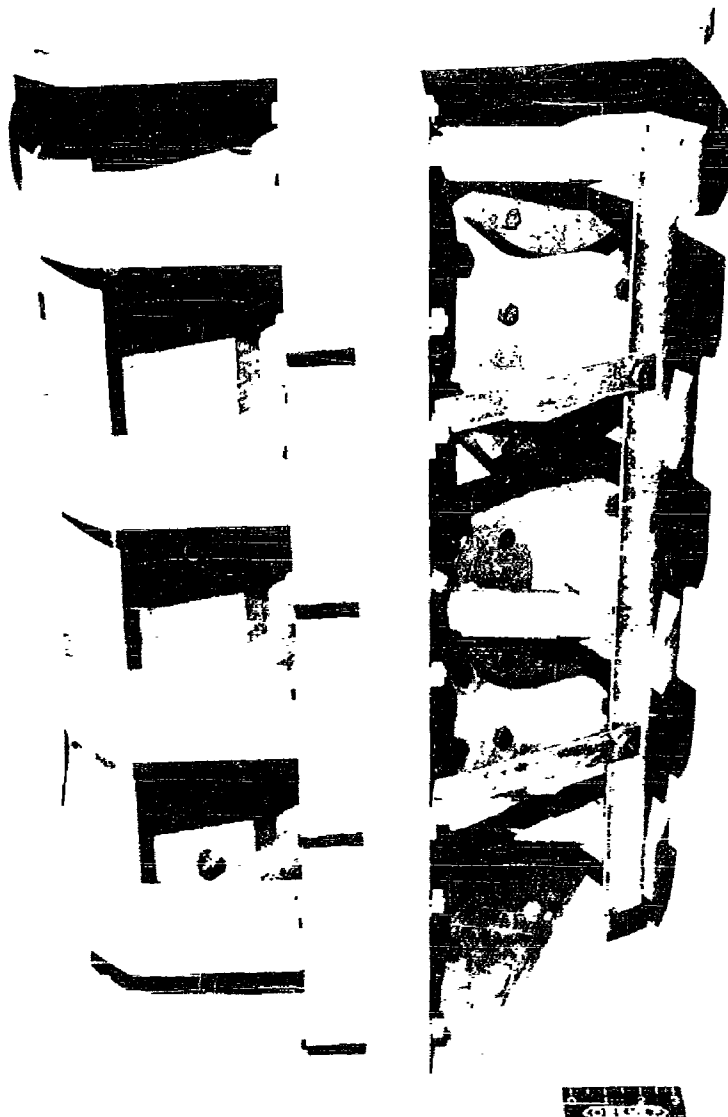


FIGURE B28
CENTRIFUGALLY CAST DISCS

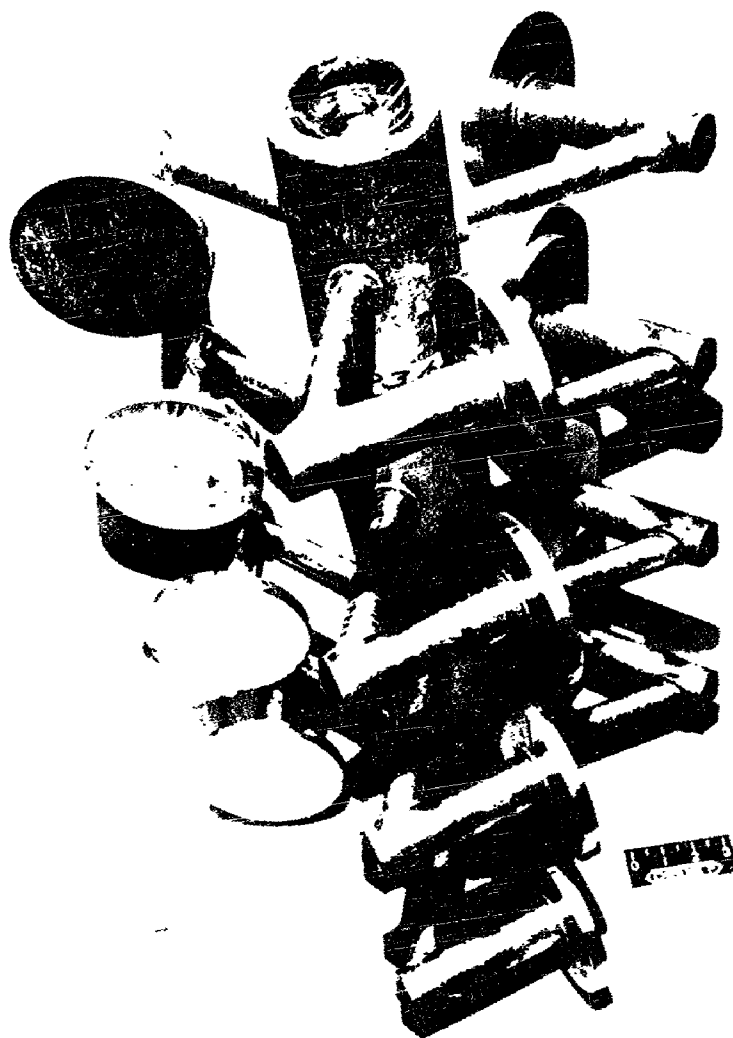
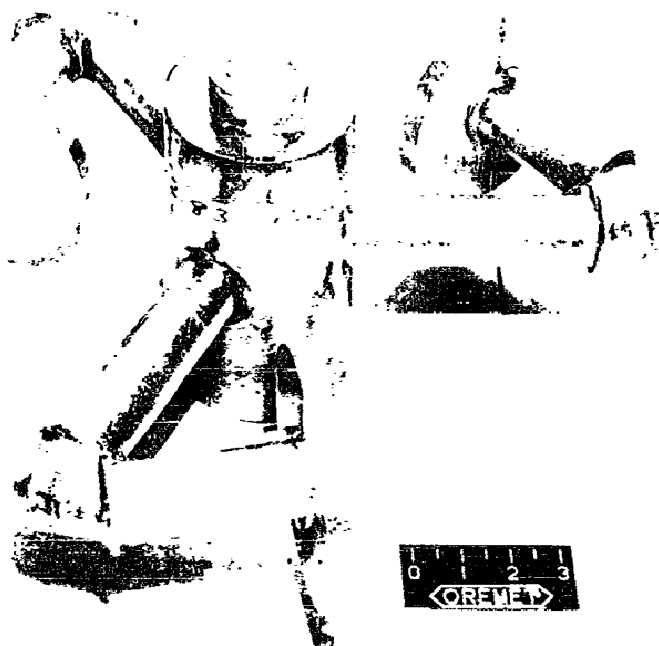


FIGURE B29

CENTRIFUGALLY CAST DISCS



SURFACE TREATMENT OF TITANIUM CASTINGS

Removal of Gates and Risers

Gates and risers were removed from the castings by oxy-acetylene torch cutting, power hack-sawing, and abrasive cut-off wheel. Of these the use of the abrasive cut-off wheel was found to be the most satisfactory. The oxy-acetylene torch cut-off increases oxygen contamination of the foundry scrap and requires excessive hand grinding to clean up the casting. Although power sawing produces a satisfactory cut, the abrasive cut-off wheel can trim closer to the casting and is much faster than sawing. The cost of the abrasive discs was compensated by lower labor cost compared to power sawing.

Grinding and belt sanding of cast titanium was used for primary cleaning. Rough grinding to remove material at gates, risers, and flash points was the most effective method of metal removal. Aluminum oxide wheels were used and are operated at approximately 3,030 surface feet per minute. All castings were hand held.

The grinding belt was used for finishing the cast titanium to remove rough grinding marks and for slight blending and flash removal. Material removal was fast and efficient, with metal removal efficiency increased by using a white-tallow base belt lubricant. Belt operating speed was 1100 surface feet per minute with average belt life of 8 hours. The belts were made of silicon carbide abrasive bonded to heavy cloth. The most efficient sanding was accomplished using a ribbed hard rubber wheel at the point of pressure contact. The ribbed-rubber wheel gives maximum life to the sanding belt coupled with efficient metal removal.

Using present methods of grinding, there has not been any evidence of grinding damage of cast surfaces. An attempt was made to damage a cast titanium specimen by poor grinding techniques. The experiment consisted of hand finishing the specimen on a grinding belt using abnormally high hand pressures. Microstructural examination of the surface did not reveal any discontinuities in the ground surface. The experiment resulted in destruction of the grinding belt.

Cleaning of Castings

The use of a shot type abrasive was not suitable for cleaning castings. Shot blasting does not adequately remove surface oxides or fused graphite. The shot provided a surface deformation action whereas titanium castings require an abrasive that will produce a cutting action to remove the surface contaminants.

Grit blasting of castings with an angular abrasive material was a satisfactory method of cleaning. The angular grit cleaning material produced a cutting action and readily removed surface oxides and graphite from titanium alloy castings. The most satisfactory cleaning material of those used has been a chilled iron blasting grit. The material continually breaks up to produce new cutting edges and as used, a finer surface finish was obtained. Adding a small quantity of new material at frequent intervals maintains the cleaning action and surface finish at acceptable standards. A detrimental effect of using the sharp cast iron material is the coating of the cleaned part with finely divided iron particles, which later rust with detriment to appearance. A second although not serious effect of the use of iron shot is possible slow build-up of the iron level in chemical analysis of the cast titanium due to use of shot blasted scrap. This effect is not serious as additions of new material dilute the iron to acceptable levels. If necessary, surface contamination by iron dust can be removed by light pickling of the scrap.

Other types of grit used were "malleabrasive" grits of various grain sizes. The malleabrasive grit is more ductile than the chilled iron grit and wears by deformation rather than by breakdown. The wearing action causes the sharp angular structure to become round and ineffective in removal of surface oxides and fused material. The material gradually performs as shot in reduced cleaning effectiveness.

Sand blasting of cast titanium has been effective in producing a fine surface finish with small metal removal rate. Three types of sands that have been used are Garnet sand, Banding sand, and Flint sand obtained from Idaho Garnet Company, Spokane, Washington, and Ottawa Silica Company, Ottawa, Illinois.

The Garnet sand is the tougher abrasive and holds up well during use. Metal removal is slow, therefore, primary cleaning of surfaces with this material is not recommended. The material used was fairly coarse and did not give a fine appearing surface finish.

The Flint sand and Banding sand provided a relatively fine surface finish. These were used as a secondary cleaning medium to remove any Iron contamination caused by the initial cleaning with the chilled Iron grit. The Flint and Banding sands have rapid breakdown characteristics and, after one hour's use the major portion of the abrasive is reduced to fines.

Abrasive tumbling has not been found an acceptable method of cleaning cast titanium. The abrasive material evaluated was Tumblex "a" No. 3, an Alundum material. The cast titanium specimens were tumbled in the abrasive media for 24 hours to determine the effectiveness in surface cleaning. At the end of 10 hours, no removal of the oxide discoloration layer on the cast surface was evident. Surface roughness was reduced at 16 hours with an acceptable product at 24 hours. The flash and burrs on the specimen were not removed during the tumbling cycle but instead had a polished appearance. It should be noted that this was a limited study and more favorable results might be obtained through more trials using other equipment and abrasives.

Chemical Removal of Surface Material

A satisfactory acid solution and processing techniques were developed for chemical removal of surface material from the castings. Fatigue tension tests were used to determine the necessity of this step and to establish the amount of material which should be removed.

The chemical removal solution developed is as follows:

Nitric Acid	15 to 25 oz./gallon
Hydrochloric Acid	5 to 9 oz./gallon
Ratio HNO ₃ / HCL	2 to 3
Free Hydrofluoric Acid	3 to 8 oz./gallon
Acetic Acid or ...	3 to 4 oz./gallon
Oxalic Acid (optional)	1 to 4 oz./gallon
Disodium Fluoroborate	1 to 2 oz./gallon
Wetting Agent (optional)	To adjust surface tension to
"Ultrawet" (optional)	1 to 4 oz./gallon
Retention Time (min)	

This solution may be maintained until 20 ounces of titanium per gallon of solution has been dissolved. It can be operated at 90°F to 130°F to adjust metal removal rate, which varies from .0003 to .0015 inches per minute over this temperature range. To maintain a reasonably constant specific removal rate, the temperature should not vary more than five degrees from nominal during the removal operation.

Before being chemically cleaned or pickled, the castings must be abrasive blasted to remove any mold particles or oxidation. During chemical removal several dimensions on the parts should be frequently spot checked for thickness control. After pickling, the castings must be water rinsed.

A series of tension-tension fatigue tests were conducted to establish the proper amount of surface removal depth. Specimens were 1/2-inch gage diameter and were tested in the following conditions:

- (a) As-cast, with normal abrasive cleaning,
- (b) As above plus chemical removal of .005-inches per surface,
- (c) Same as (a) plus chemical removal of .010-inches per surface,
- (d) Same as (a) plus chemical removal of .015-inches per surface.

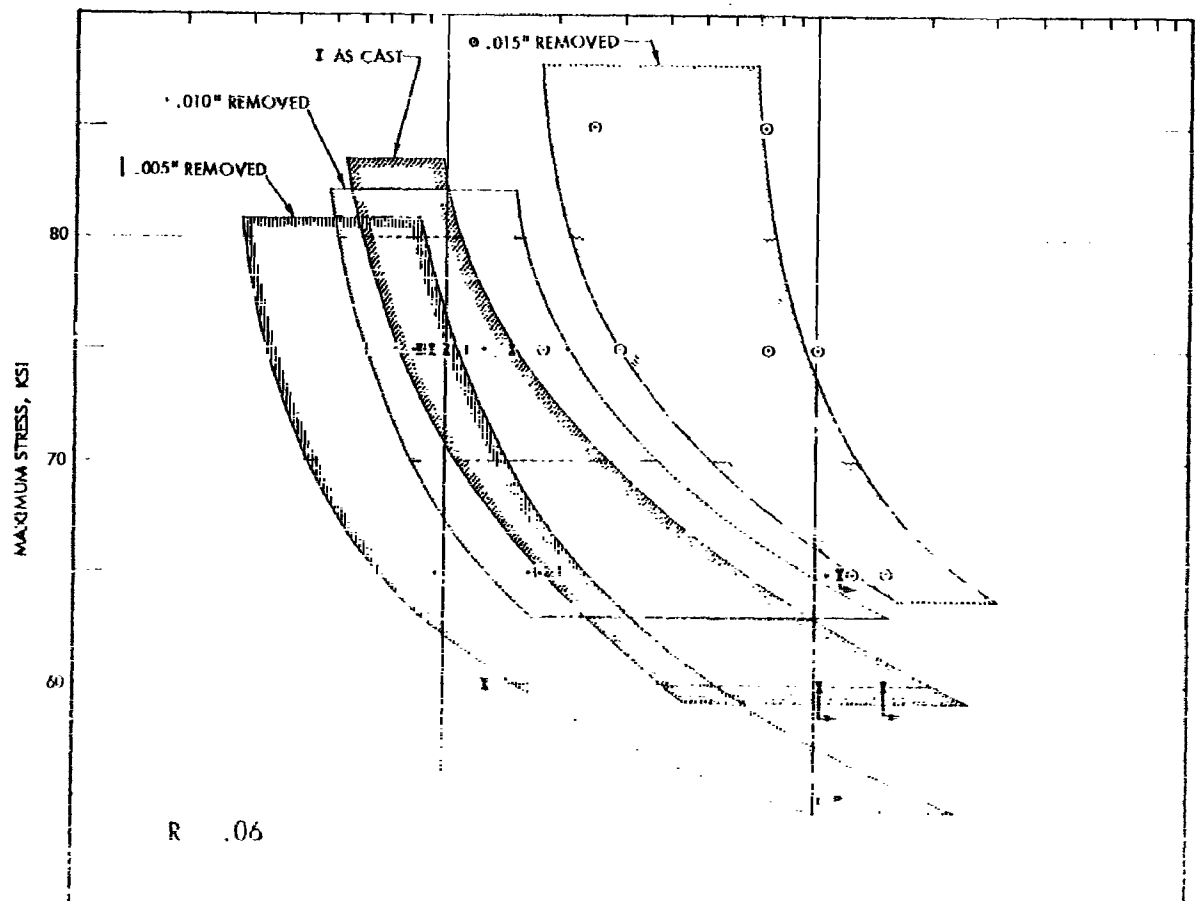
The specimens were tested in a conventional Sonntag SF - 10U universal fatigue testing machine at 1800 cycles per minute and a stress ratio of $R = .06$. The specimens were all from the same heat, were x-ray inspected, and the acceptable specimens selected to randomize quality.

The results of the fatigue tests are tabulated in Table J16 and illustrated by S - N curves in Figure B3Q. In general, removing only .005 inches per surface damaged fatigue life, removing .010 inches per surface restored the fatigue life to that of the as-cast condition, and removal of .015 inches per surface provided significant improvement over the as-cast condition.

It is recommended that, in titanium alloy casting applications requiring best fatigue properties, .015-inches per surface be chemically removed using the solution and process described.

FIGURE B30

FATIGUE PROPERTIES OF AS-CAST AND PICKLED Ti-6Al-4V CAST SPECIMENS



SECTION C

ALLOY DEVELOPMENT

Development of Ti-6Al-4V Alloy

Early in this program, it was necessary to choose an alloy for use in casting trials, mold development trials, and other work which could not be delayed until alloy development studies were completed. Because of the general acceptance of the 6Al-4V titanium alloy for wrought products, that alloy was chosen for these studies with the intention of substituting a more suitable casting alloy when one was found. As discussed in the portion of this report which discusses general alloy development, a casting alloy significantly better than Ti-6Al-4V was not found.

A series of approximately 440 heats of Ti-6Al-4V was studied to establish effect of analysis on mechanical properties, practical analysis limits, practical guaranteed minimum mechanical properties, and correlations of processing variables with resulting interstitial levels.

First, frequency distribution bar charts (histograms) were prepared to show the distributions of oxygen, nitrogen, hydrogen, carbon, iron, aluminum, and vanadium which were characteristic of the equipment and practice developed in this program. The histograms are presented as Figures C1 through C7. Much of the early work in the program was done without concentrated effort to maintain close control of chemical analysis, since other variables received the main focus of attention. As a result, a fairly wide spread of analyses was obtained which proved of benefit in studying effects of analysis on properties.

As shown in Figure C1, the oxygen content varied from .08 to .37 per cent and averaged approximately .21 per cent. This high oxygen range is the result of recycling of foundry scrap, since from .02 to .05 per cent oxygen is picked up during each melt and recycling of material from previous heats compounds the pick up. Because of the economic necessity to use foundry and other titanium alloy scrap in the casting process, high interstitial oxygen contamination has been a problem. To maintain oxygen at acceptable levels, it is necessary to occasionally discard a portion of the normal foundry scrap (gates, risers, etc.) and dilute the oxygen content by adding low-oxygen sponge or electrode stock to the heat. Therefore, the amount of oxygen that can be tolerated as a maximum in titanium alloy castings has a significant bearing on the cost of the final product.

For study of effects of each specific element on mechanical properties, the analysis ranges of the other alloying and interstitial elements were restricted to closer-than-normal tolerances, to minimize obscuring the effects of the element in question by effects of scatter of the other elements. In the case of oxygen, only those heats were used which were within the following analysis limits:

Nitrogen	.021 - .032 per cent
Carbon	.10 per cent maximum
Hydrogen	.010 per cent maximum
Aluminum	5.8 - 6.2 per cent
Vanadium	3.8 - 4.2 per cent

This left 135 heats within these analysis limits having mechanical properties available, with oxygen ranging from .07 to .37 per cent. Oxygen was then plotted against ultimate tensile strength (Figure C8) and against reduction of area (Figure C9) as scatter diagrams. As shown in the figures, increased oxygen increases strength with fair correlation and decreases ductility with considerable scatter in individual values. As also may be noted in the scatter diagram for oxygen versus reduction of area, average reduction of area decreases as oxygen is increased, but the minimum reduction of area does not appear to decrease in the range of oxygen covered (i.e. up to about .25 per cent oxygen). Next, to establish the allowable production maximum for oxygen, all heats which fell in the normal analysis range (Al 5.5 - 6.5, V 3.5 - 4.5, N .07 max., C .10 max., H .015 max., Fe .30 max.) except for oxygen were separated into groups according to oxygen content, as follows:

Group A	less than .16 per cent oxygen
Group B	.16 - .21 per cent oxygen
Group C	.21 - .26 per cent oxygen
Group D	.26 - .31 per cent oxygen

Histograms for ultimate strength, elongation, and reduction of area were then plotted for these groups and are presented as Figures C10, C11, C12 and C13. Averages and standard deviations (σ) were computed for each curve using the conventional statistical analysis method for the "normal" frequency curve. (It should be noted that the histograms for mechanical properties are not "normal" distributions but are skewed slightly toward the high-value ends of the curves, resulting in conservative minimum properties since the standard deviation is slightly enlarged by the high values which are not undesirable except for this influence). Two standard

deviations were subtracted from the average values for each curve to establish "minimums" for this study. Design allowable minimums are discussed later in this report. This analysis method is explained in the appendix. As is shown by the histograms and the scatter diagrams, the average and minimum values for ultimate strength are increased as oxygen increases, but for both elongation and reduction of area the averages decrease but the minimum values either remain fairly constant or actually increase due to lesser amounts of scatter in all properties as oxygen is increased. As a result it is advantageous to allow .25 per cent as a maximum for oxygen in titanium castings compared to the .20 per cent maximum usually allowed in wrought products. The minimum ultimate strength is increased without loss in ductility. Consequently, .25 per cent has been established as a maximum for oxygen in the specification for Ti-6Al-4V castings.

The scatter diagrams for nitrogen show that it strengthens Ti-6Al-4V considerably without detriment to ductility, and that the correlation between analysis and properties is better for nitrogen than for oxygen. It may be desirable to establish a minimum as well as the present maximum analysis value for nitrogen to take advantage of its strengthening effect. This was not done since nitrogen independently is difficult to control but as is shown in Figure C16, tends to be high when oxygen is high. Since recycling of foundry scrap is desirable, oxygen content will normally be near the high limiting value, and nitrogen will be also maintained at a relatively high value. As a result, the minimum design values established for Ti-6Al-4V during this program may possibly be increased as more production experience is gained with heats approaching the higher analysis limits for oxygen and nitrogen.

The scatter diagrams and analysis comparisons for aluminum and vanadium show negligible effect of analysis on strength or ductility for the composition ranges typical of the process. For these histograms, ranges for the other elements were selected to the following limits:

for study of aluminum,

Vanadium	4.8 - 4.2 per cent
Oxygen	.13 - .25 per cent
Carbon	0 - .10 per cent
Hydrogen	0 - .01 per cent

for study of vanadium,

Aluminum	5.8 - 6.2 per cent
Oxygen	.13 - .26 per cent
Carbon	0 - .10 per cent
Hydrogen	0 - .01 per cent

The study of carbon showed that increasing carbon provides slight strengthening effect with no significant effect on ductility over the normal analysis range.

Considering past experience with wrought alloys, the normal analysis distributions characteristic of this process, the economics of using foundry and other scrap in the casting process, and the influence of analysis on mechanical properties, the following analysis limits were established as controls for production castings of the Ti-6Al-4V alloy:

Aluminum	5.5 - 6.5 per cent
Vanadium	3.5 - 4.5 per cent
Oxygen	.25 per cent maximum
Nitrogen	.07 per cent maximum
Carbon	.10 per cent maximum
Iron	.30 per cent maximum
Hydrogen	.015 per cent maximum.

A brief study of the effect of grain size of cast Ti-6Al-4V as affected by cooling rate during solidification consisted of pouring a heat of Ti-6Al-4V into four tensile coupon molds, made of rammed graphite, shell graphite, machined graphite, and copper. A macro-photograph of the sectioned coupons is shown as Figure C17. The properties of tensile tests of these coupons were as follows:

Mold Material	U.T.S. psi	Y.T.S. (.2%) psi	Elong., %	R.A., %	Notch * Tensile Strength
Machined Graphite	135,000	116,000	7.6	13.0	205,000
Rammed Graphite	135,000	115,000	7.6	13.0	194,000
Shell Graphite	137,600	120,000	5.5	10.1	198,000
Copper	137,500	118,000	7.1	8.3	205,000

* $K_t = 3.37$ Heat No. 2213 - F251

It was concluded that the various molds had insignificant effect on mechanical properties.

The as-cast properties of the heats of Ti-6Al-4V are tabulated in Table J17.

Elevated temperature properties of as-cast Ti-6Al-4V were determined and presented in Table J18 and Figure C18.

Several attempts were made to improve the strength and ductility of the Ti-6Al-4V alloy by heat treatments. The first trials were attempts to modify the cast structure by cycling through the beta-alpha transformation temperature. Next, attempts were made to homogenize the structure by longer solution times than are used for forgings. Neither of these treatments produced improvement over conventional heat treatments. In general, greater strength with concomitant lower ductility was obtained at higher solution temperatures and lower aging temperatures. As solution temperatures were progressively reduced to 1550° F and aging temperatures were increased to 1100° F in attempting to obtain better ductility, ultimate strength increase over the as-cast condition was in the order of 15,000 psi with an accompanying loss of ductility to about one half the as-cast ductility values. Furnace cooling from 1500° F to 1000° F followed by air cooling resulted in reduced ductility with no increase in strength, as compared to as-cast properties.

Further attempts to improve the ductility of cast Ti-6Al-4V consisted of annealing for two to eight hours at 1000°F to 1200°F. Although the results were erratic, the annealing at 1150°F was generally slightly beneficial to ductility without affecting strength. At the other temperatures tried, no advantage in annealing was found. If annealing of Ti-6Al-4V castings is desired as a stress relief or for other reasons, the 1150°F treatment is recommended. The carbon, oxygen, nitrogen, iron, and chromium analyses of the heats used for the annealing treatment investigation were studied to determine if there was a correlation between the amounts these elements and the tendency for ductility to be increased or decreased by annealing. Such correlation was not found.

A heat treatment was developed which simultaneously increased strength and ductility by significant amounts, however the heat treatment was considered too difficult or impossible to accomplish on cast shapes.

The heat treatment consisted of holding at 1900°F for 30 minutes, quenching to 1250°F in less than six seconds, hold at 1250°F for 8 hours, air cool to room temperature, heat to 1500°F and hold for 2 hours, furnace cool at not more than 5F per minute to 1000, and air cool. This treatment provided approximately 3000 psi strength increase with an increase of reduction of area of 5 per cent. A further solution and aging treatment consisting of 1650°F for 20 minutes, water quench, 1000°F for 4 hours, and air cool resulted in a strength increase over as-cast of 19,000 psi with reduction of area loss of 3.5 per cent.

The results of all heat treatment trials on the Ti-6Al-4V alloy are in Table J19.

Development of Casting Alloys Other Than Ti-6Al-4V

The purpose of this portion of the work was to develop titanium casting alloys with properties superior to those of Ti-6Al-4V.

The as-cast mechanical properties of the experimental alloys are in Table J20.

Since the experimental alloys were produced from new sponge and high purity alloying elements, the carbon, oxygen, and nitrogen contents are generally lower than normally present in the Ti-6Al-4V heats which were produced from recycled foundry returns. Consequently, in some instances the experimental alloys are of lower strength and higher ductility than would be expected if they were used in normal production.

Several elements that can be alloyed with titanium to strengthen the alpha phase without producing the beta phase in the room-temperature microstructure are: aluminum, silicon, carbon, oxygen, nitrogen, tin, and zirconium. Since alpha alloys do not respond to heat-treatment, they have good weldability. Also because of the relatively small temperature difference between the liquidus and solidus of alloys of titanium and these elements, it would be expected that these alloys would have good fluidity in the sense of ability to run a thin section without cold shuts or misruns; and also have good feeding characteristics in the sense of enabling risers to feed relatively long distances without shrinkage porosity. Hence, it was hoped that an alpha alloy would be found useful as a low-strength weldable alloy of good castability.

Elements known to be soluble in beta titanium and which tend to make the beta phase stable and strengthen it are: vanadium, manganese, chromium, iron, columbium, molybdenum, nickel, and hydrogen.

Alloys containing manganese, chromium, iron, and nickel undergo a eutectoid transformation from the beta phase to alpha and form inter-metallic compounds as the alloys are cooled to room temperature. In general, as the rate of transformation increases, the transformation temperature and the alloy content of the eutectoid composition decreases for alloying elements toward the right side of the periodic table and having greater differences in atomic diameter as compared to titanium. With the elements vanadium, columbium, and molybdenum, which are not as far to the right of titanium in the periodic table and have atomic diameters more nearly equal to that of titanium, the eutectoid transformation has not been found to occur.

The 5Al, 2 1/2Sn alloy and the 8Al, 2Cb, 1Ta alloys were tried because they were well-known standard alpha alloys at the time.

The 6Al and 8Al alloys were tried because investigations by others had indicated that aluminum strengthens titanium in additions up to approximately 8 per cent.

The 2Cu and the 6Al - 2Cu alloys were tried because the equilibrium diagram for Ti-Cu indicates that Ti₂Cu can be precipitated from alpha titanium. It was hoped that age hardening could be obtained in these alloys and that in the second alloy, additional solid solution hardening would be obtained. The 3Mo - 1/2Be alloy was tried for this same reason.

The 6Al - 1Si alloy was tried in a follow-on of work performed at Watertown Arsenal.

Where a high density, relatively brittle alloy is desired, Ti-2Cu (alloy 41) appears to have properties superior to unalloyed titanium.

To stabilize the beta phase sufficiently so that no alpha is present at room temperature, approximately 15 per cent vanadium or 12 per cent molybdenum is required. The Ti-15V and Ti-12Mo alloys were first to detect the existence of the beta phase, but the composition of the other Ti-12V, Ti-12Mo, Ti-12V-12Mo, and Ti-12V-12Mo-12Al alloys were not for the purpose of stabilizing the beta phase.

these relatively large additions, it was believed that vanadium would be preferred to molybdenum because of density considerations.

To gain additional hardening by means of the eutectoid reaction, a Ti-15Mo-6V-2Ni alloy was cast. In an attempt to speed up the eutectoid reaction rate, a Ti-5Mo-6V-2Ni-2Al alloy was cast and tested.

To investigate the possibility of decreasing the required aging time of the B 120 VCA type of alloy by changes in the aluminum content, the following alloys were cast:

Ti-13V-11Cr
Ti-13V-11Cr-1.5Al
Ti-13V-11Cr-2.5Al
Ti-13V-11Cr-4Al

Considering as-cast properties only, the following alloys had properties similar to Ti-6Al-4V:

5Al-2.5Sn (Alloy 6)
13V-11Cr-1.5Al (Alloy 11)
13V-11Cr-2.5Al (Alloy 12)
13V-11Cr-4Al (Alloy 13)
5.5Al-5.5V-.5Fe-2Sn-.25Cu (Alloy 16)
7Al-3Mo (Alloy 19)
6.5Al-3.5Mo-1V (Alloy 29)
6Al-1Si (Alloy 47)
6Al-6V-2Sn (Alloy 49)
2Al-9Zr-13Sn (Alloy 50)

The following alloys could be expected to be in the same properties class as Ti-6Al-4V, if at the normal interstitial levels:

4Al-3Mo-1V (Alloy 27)
8Al-8Zr-.5Cb-.5Ta (Alloy 45)
6Al-1Si (Alloy 47)
6Al-4V-.5W (Alloy 55)
6Al-4V-.5Ta (Alloy 56)

Since these alloys did not appear to be superior to the 6Al-4V alloy, work on these alloys was discontinued. It appears that the alloys of the following types may be superior to Ti-6Al-4V:

4Al-4Sn-8Zr-1Fe-1V-1Cr (Alloy 60)
 4Al-4Sn-8Zr (Alloys 59 and 64)
 4Al-4Sn-8Zr-1.5Fe-1.5V-1.5Cr (Alloy 105)
 4Al-4Sn-8Zr-.5Fe-.5V-.5Cr (Alloy 106)
 4Al-4Sn-8Zr-2V (Alloy 108)
 4Al-4Sn-10Zr-2V (Alloy 110)
 4Al-6Sn-8Zr-2V (Alloy 111)
 6Al-4Sn-8Zr-2V (Alloy 112)

Additional study will be required to determine the best range of elements for this type of alloy.

The results of mechanical property tests on heat treated experimental alloys are given in Table J21.

Most of the beta and alpha-beta alloys could be strengthened by solution treat and age type of heat treatments. However the increase in strength was accompanied by loss in ductility. In general, at any given strength level an as-cast alloy can be found that is as strong and is more ductile than an alloy that was strengthened by heat treatment. Therefore, hardening heat treatments are not recommended for the cast titanium alloys.

Near the end of the program, alloys which appeared to be good candidates were further screened by a heat treatment to test "thermal stability". This consisted of heating for up to 80 hours at temperatures from 600 to 1000°F. The following were embrittled by these treatments and were eliminated from further consideration because they would not be suitable for service in this temperature range:

13V-11Cr-1.5Al (Alloy 11)
 8V-5Fe-1Al (Alloy 15)
 4Al-8Sn-8Zr-1Fe-1V-1Cr (Alloy 60)
 4Al-4Sn-8Zr-1.5Fe-1.5V-1.5Cr (Alloy 105)
 4Al-4Sn-8Zr-.5Fe-.5V-.5Cr (Alloy 106)

The 6Al-4V, 20Cr-10Al-4V-1Fe, 4Al-4Sn-8Zr-1Fe-1V-1Cr (Alloy 60), and 4Al-4Sn-8Zr-2V (Alloy 108) alloys were not embrittled by these heat treatments. Alloy 108 is strengthened by heat treatment and is stronger than the 6Al-4V alloy.

The chemical analyses of all heats in this program are given in Table J22.

The following conclusions were reached as results of the alloy development studies:

- (a) At the present stage of development, unalloyed titanium and Ti-6Al-4V are suitable for "standard" casting alloys.
- (b) Ti-2Cu has ductility superior to unalloyed titanium which has been hardened to the same strength level through interstitial additions.
- (c) The properties of an alloy of titanium containing four to six per cent aluminum, four to six per cent tin, eight to ten per cent zirconium, and two to four per cent vanadium are slightly better than those of Ti-6Al-4V. Additional work is needed to establish the best composition and interstitial levels of this type of alloy.

FIGURE C1

OXYGEN ANALYSIS DISTRIBUTION IN CAST Ti-6Al-4V

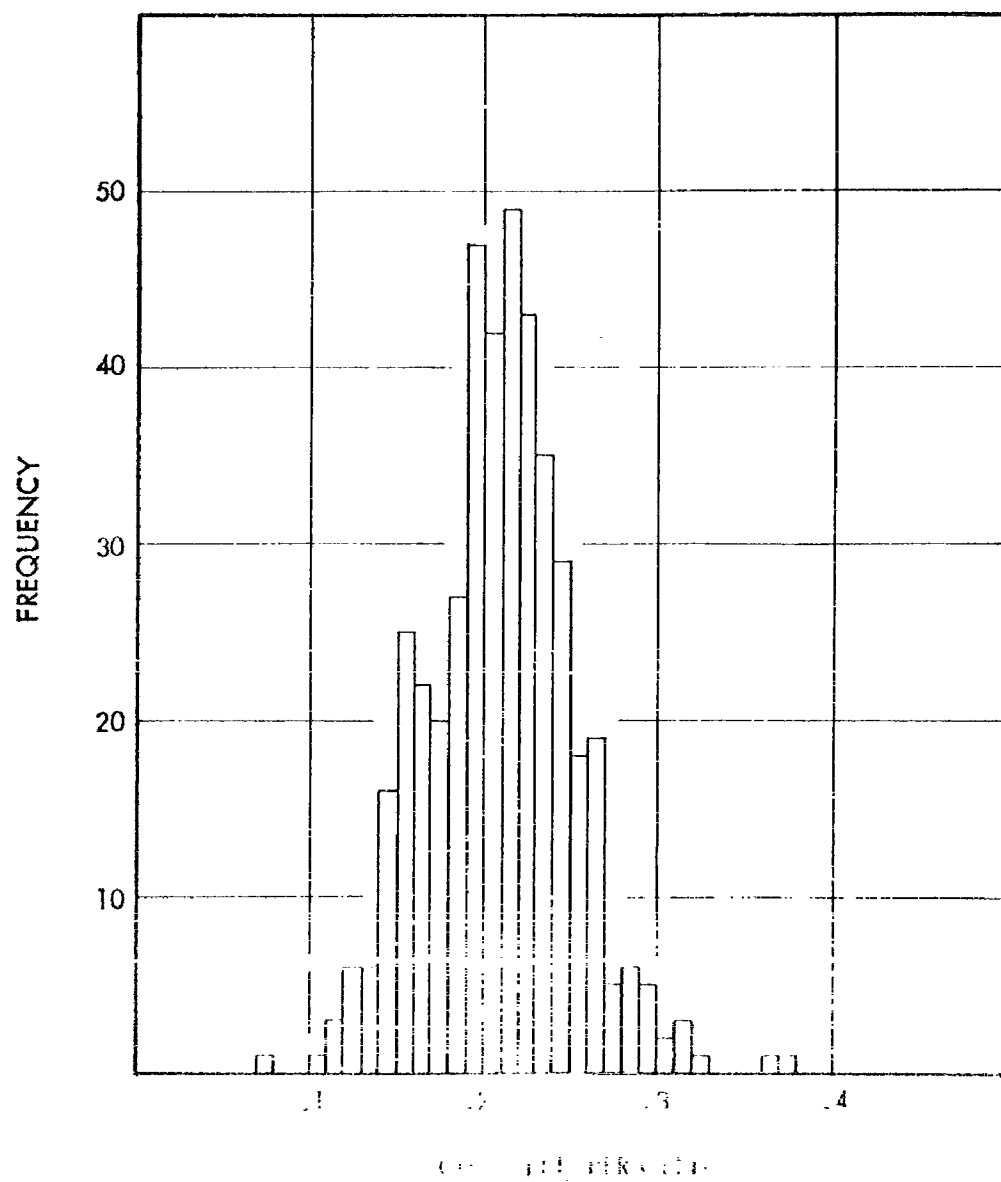


FIGURE C2

NITROGEN ANALYSIS DISTRIBUTION IN CAST Ti-6Al-4V

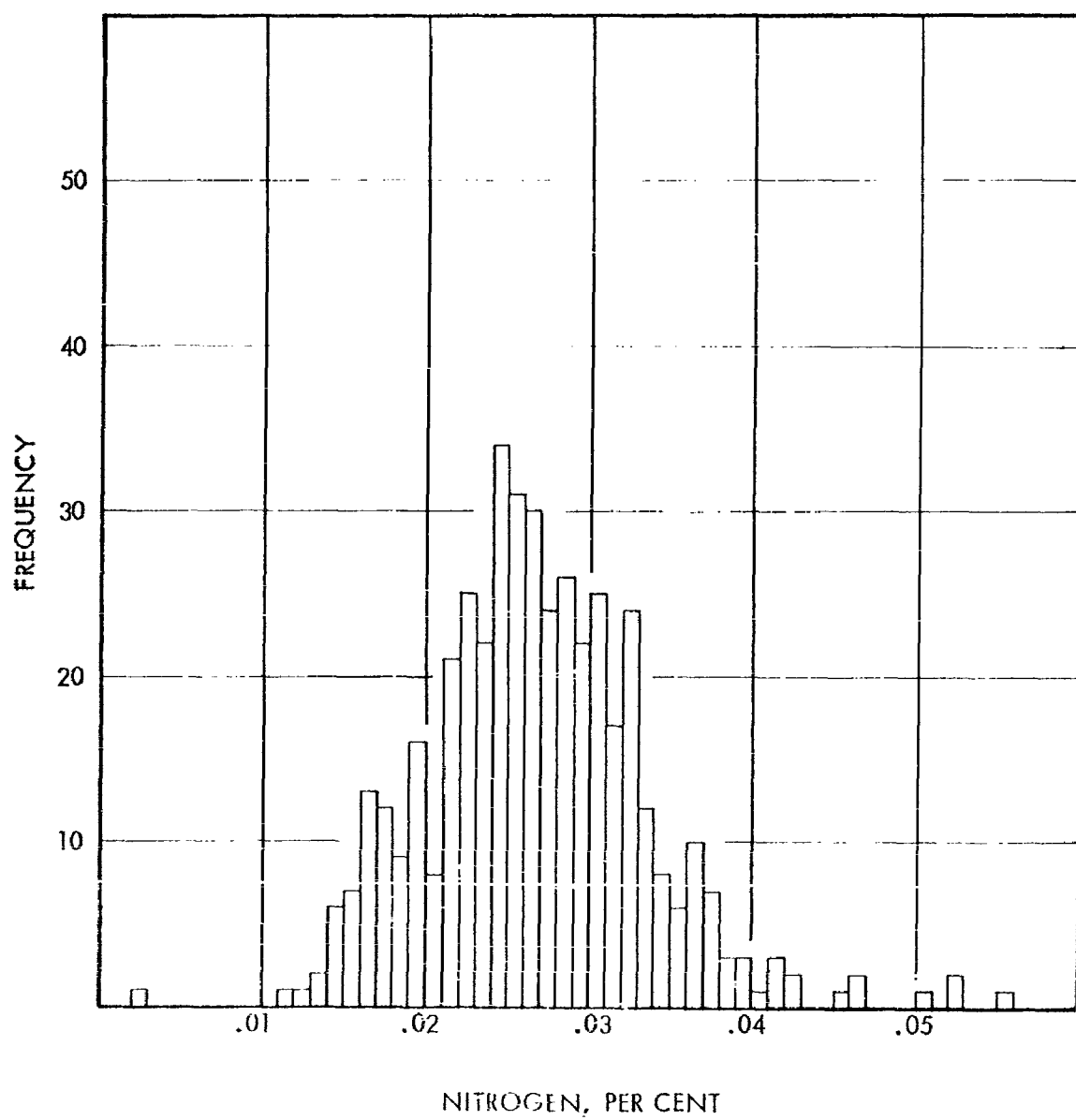


FIGURE C3

HYDROGEN ANALYSIS DISTRIBUTION IN CAST Ti-6Al-4V

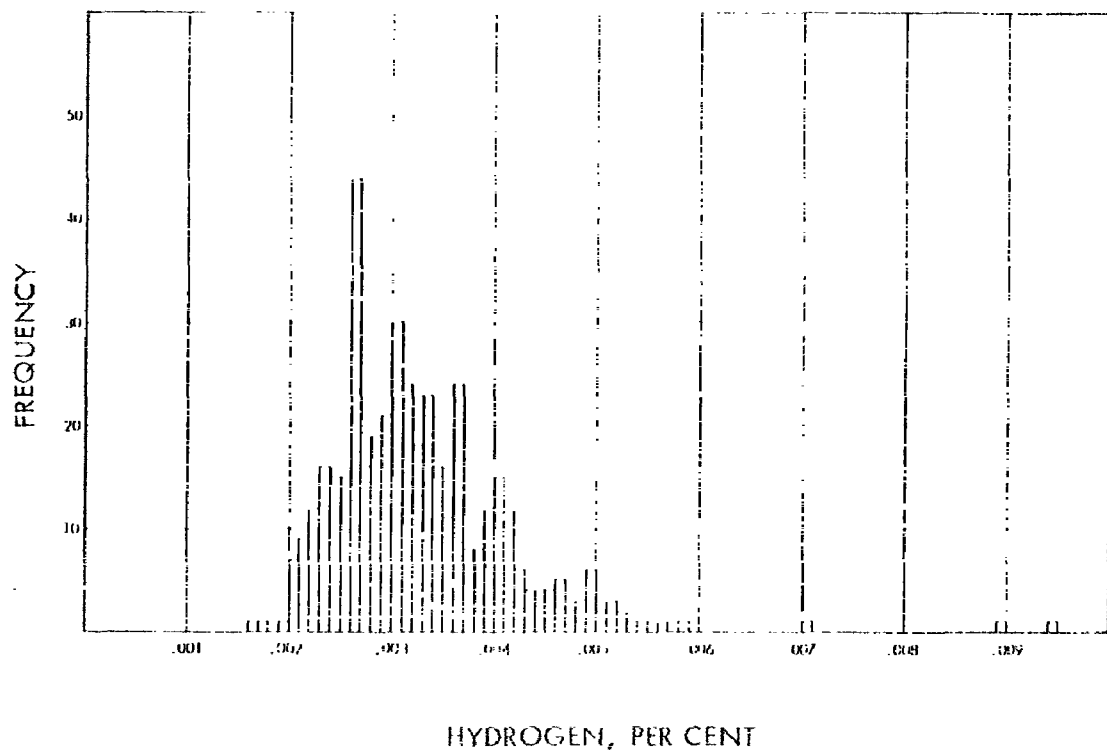


FIGURE C4

CARBON ANALYSIS DISTRIBUTION IN CAST Ti-6Al-4V

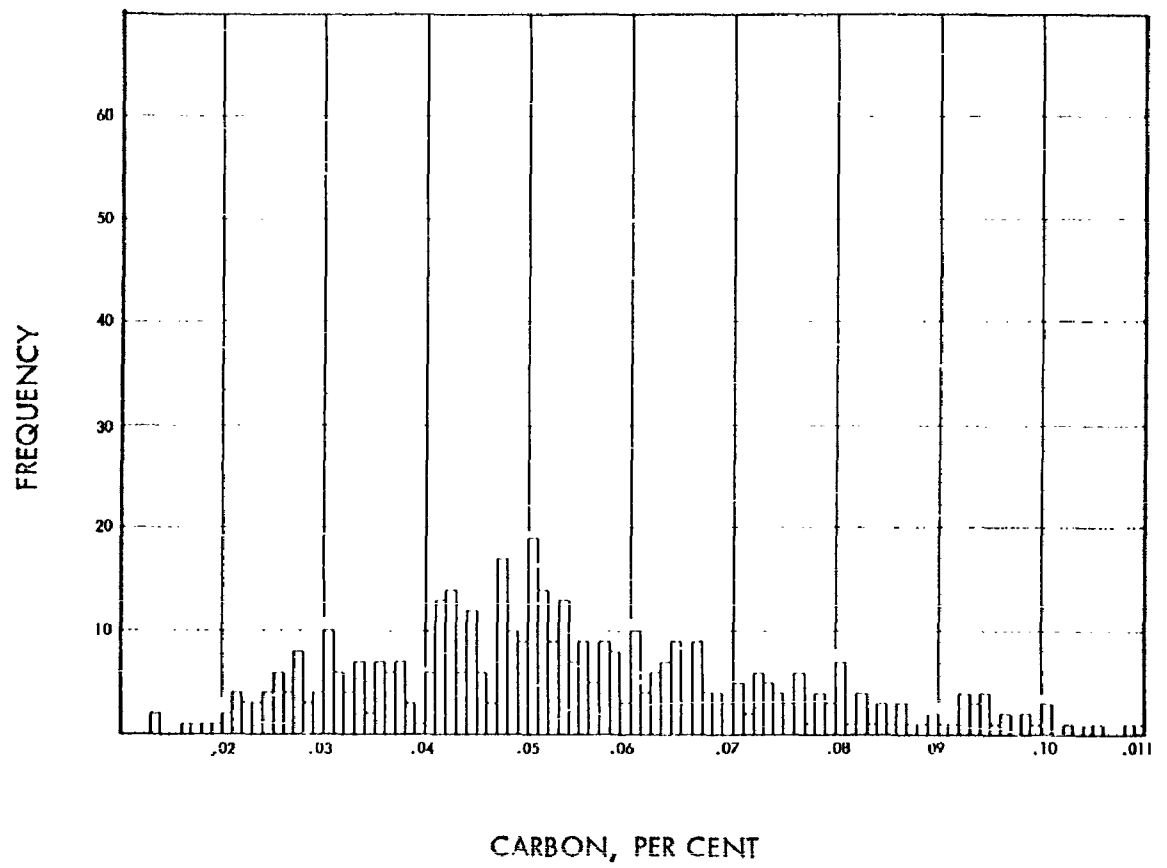


FIGURE C5

IRON ANALYSIS DISTRIBUTION IN CAST Ti-6Al-4V

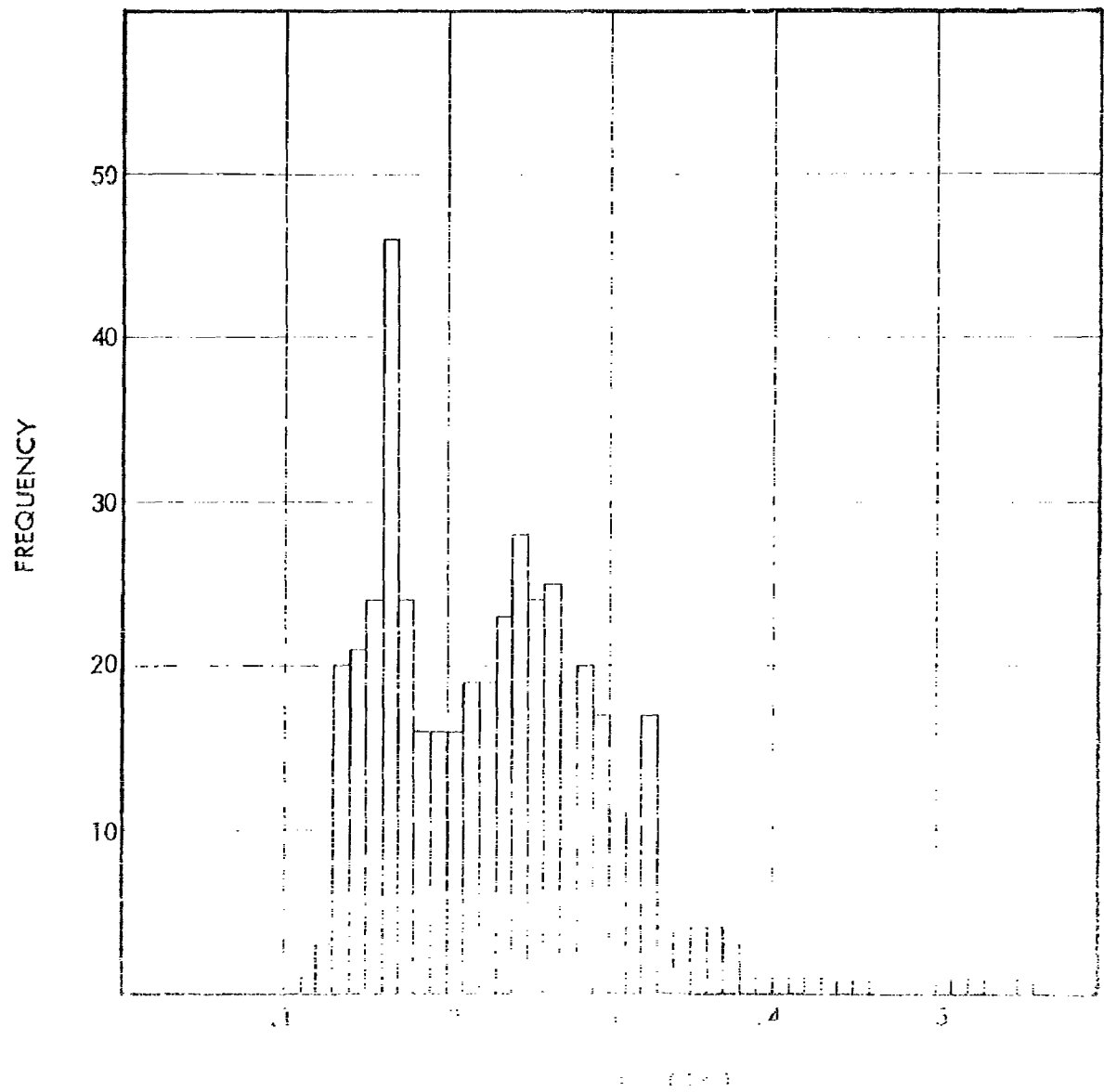


FIGURE C6

ALUMINUM ANALYSIS DISTRIBUTION IN CAST Ti-6Al-4V

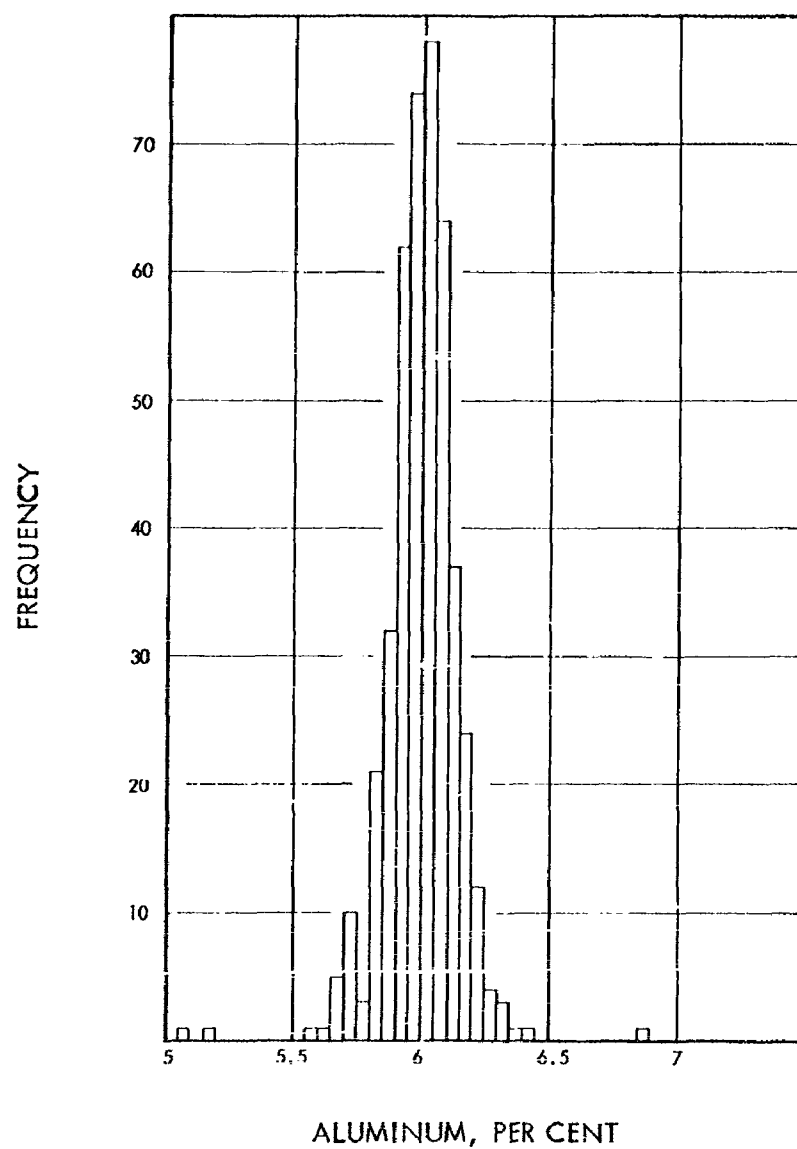


FIGURE C7

VANADIUM ANALYSIS DISTRIBUTION IN CAST TI-6Al-4V

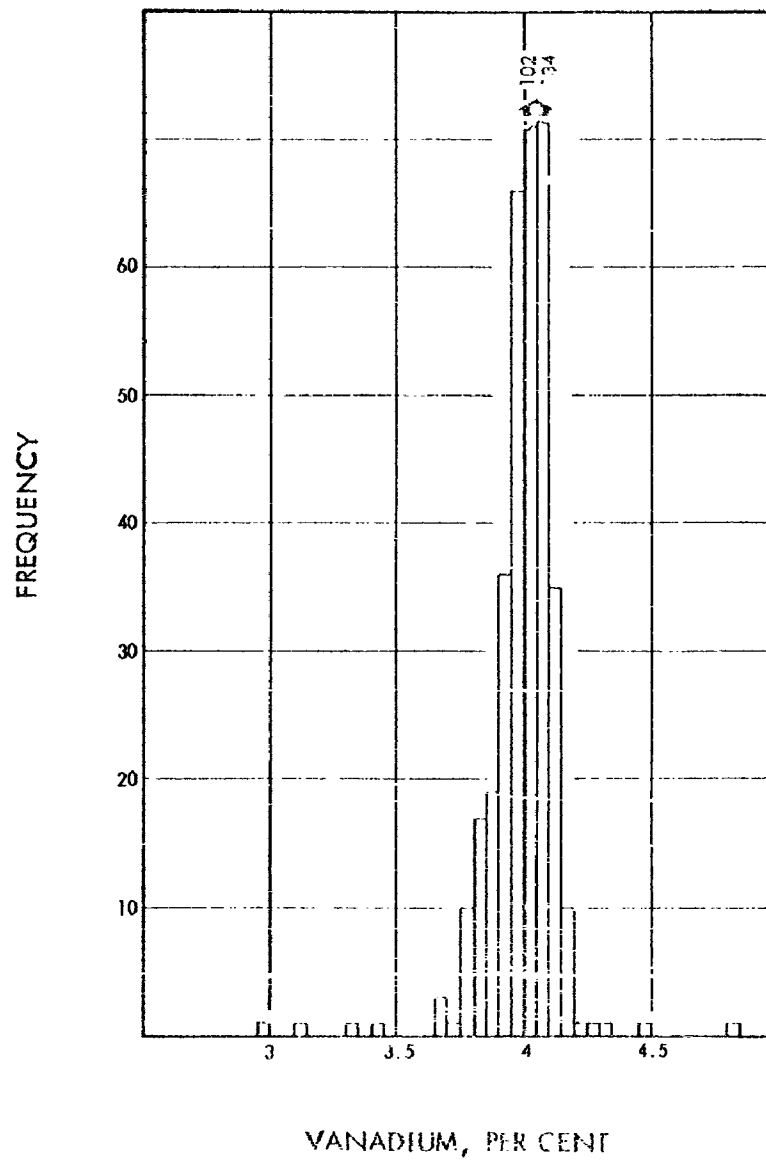


FIGURE C8

SCATTER DIAGRAM - OXYGEN VS. ULTIMATE STRENGTH

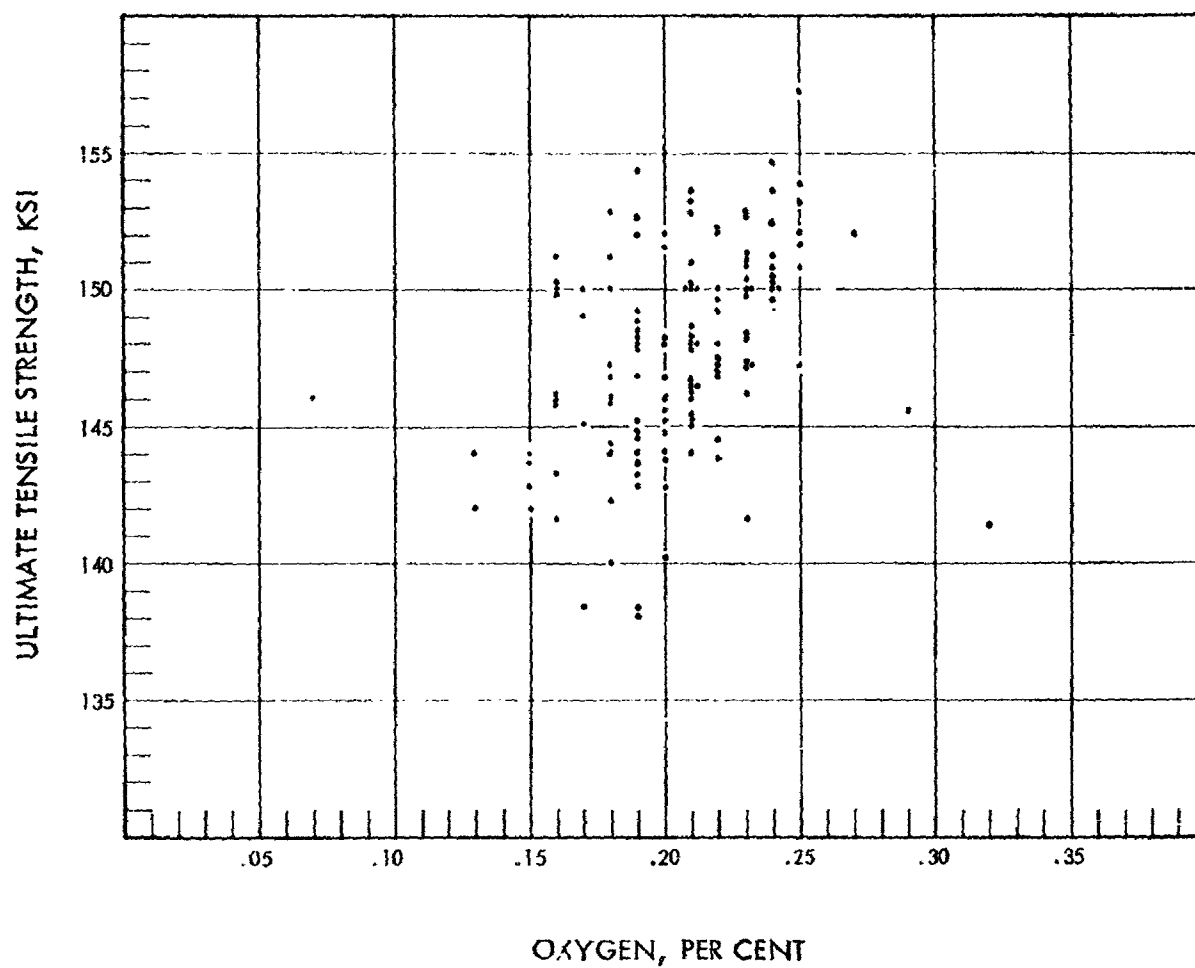


FIGURE C9

SCATTER DIAGRAM - OXYGEN VS. REDUCTION OF AREA

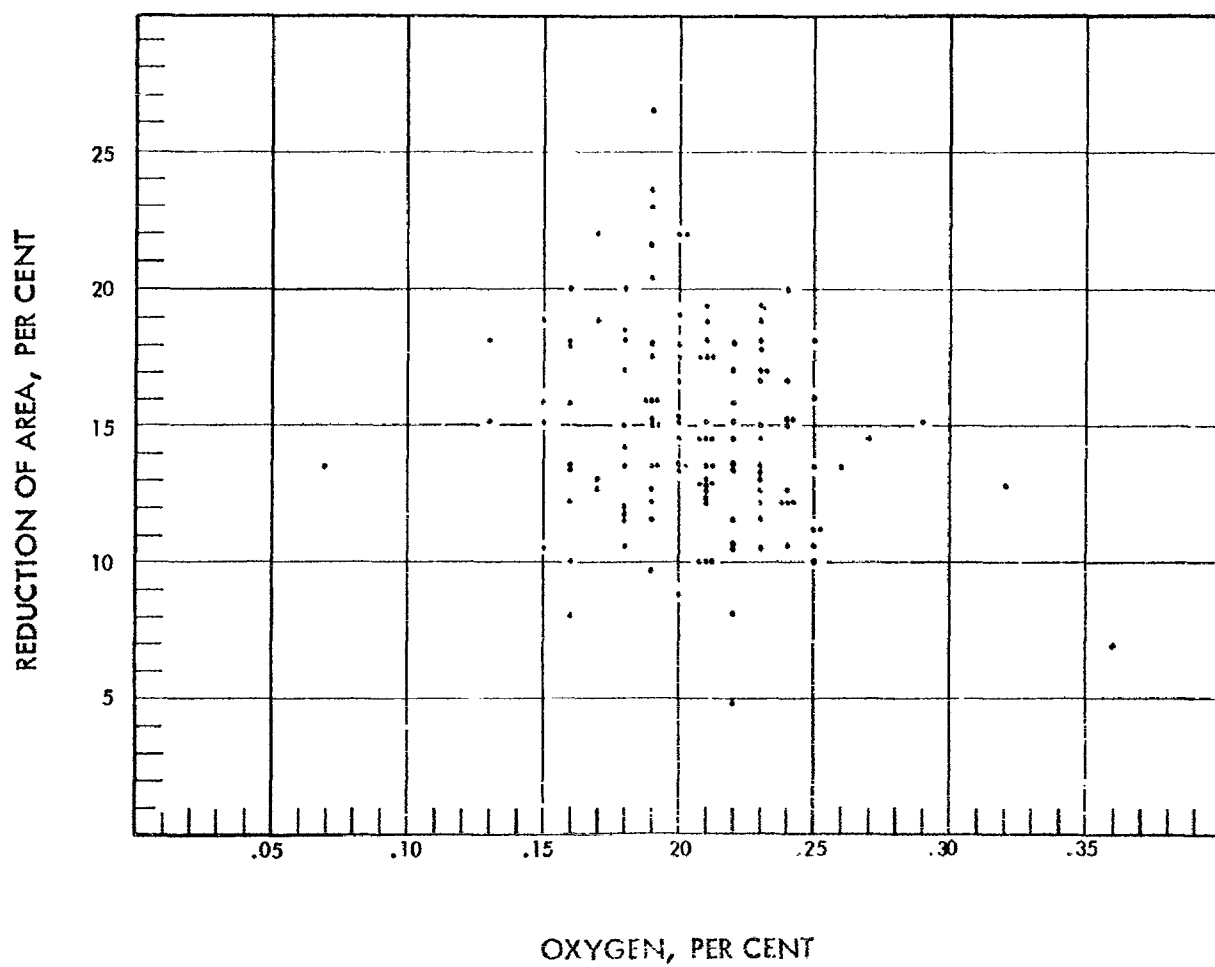


FIGURE C10

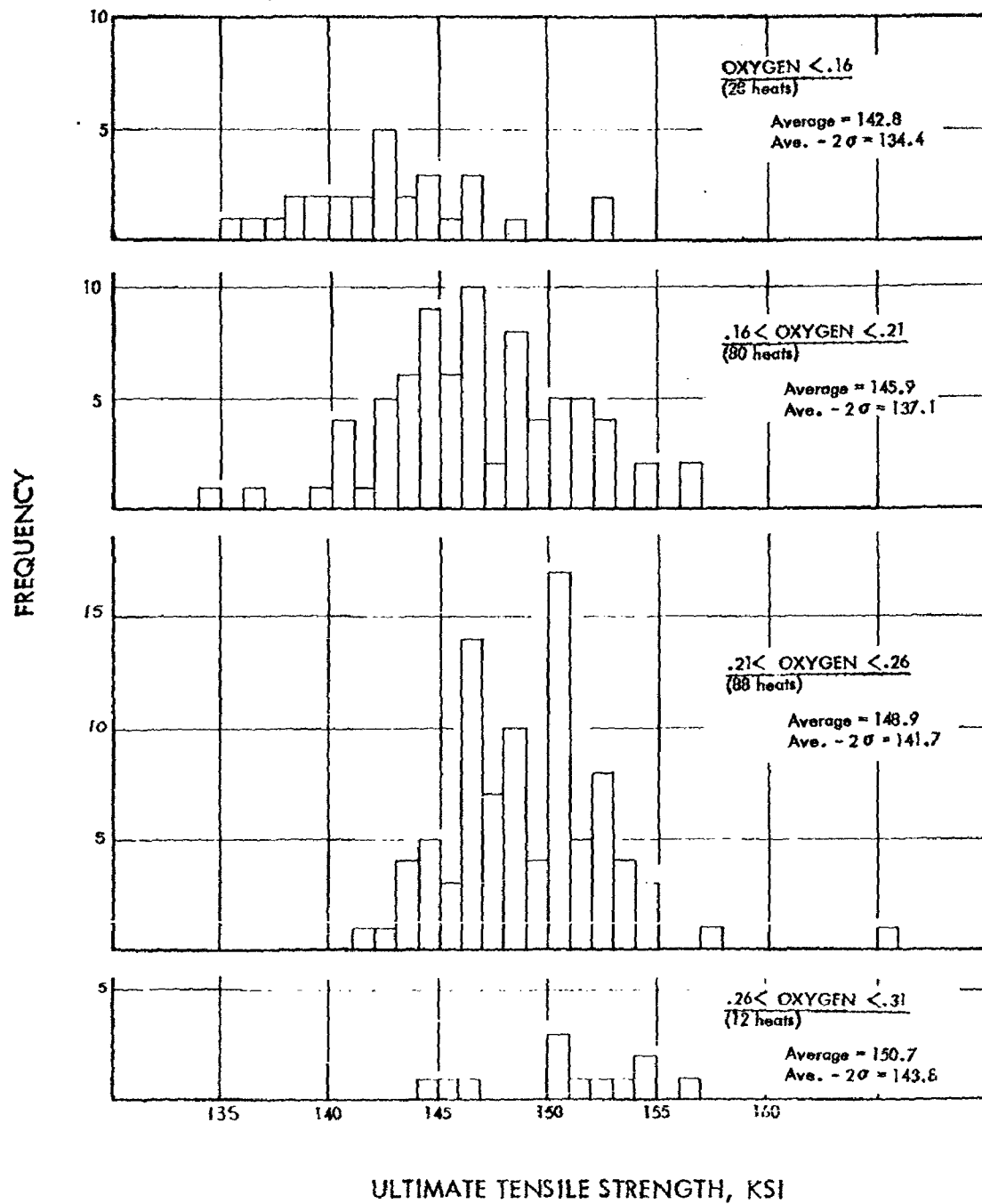
FREQUENCY DISTRIBUTIONS FOR ULTIMATE
STRENGTH FOR VARIOUS OXYGEN RANGES

FIGURE C11

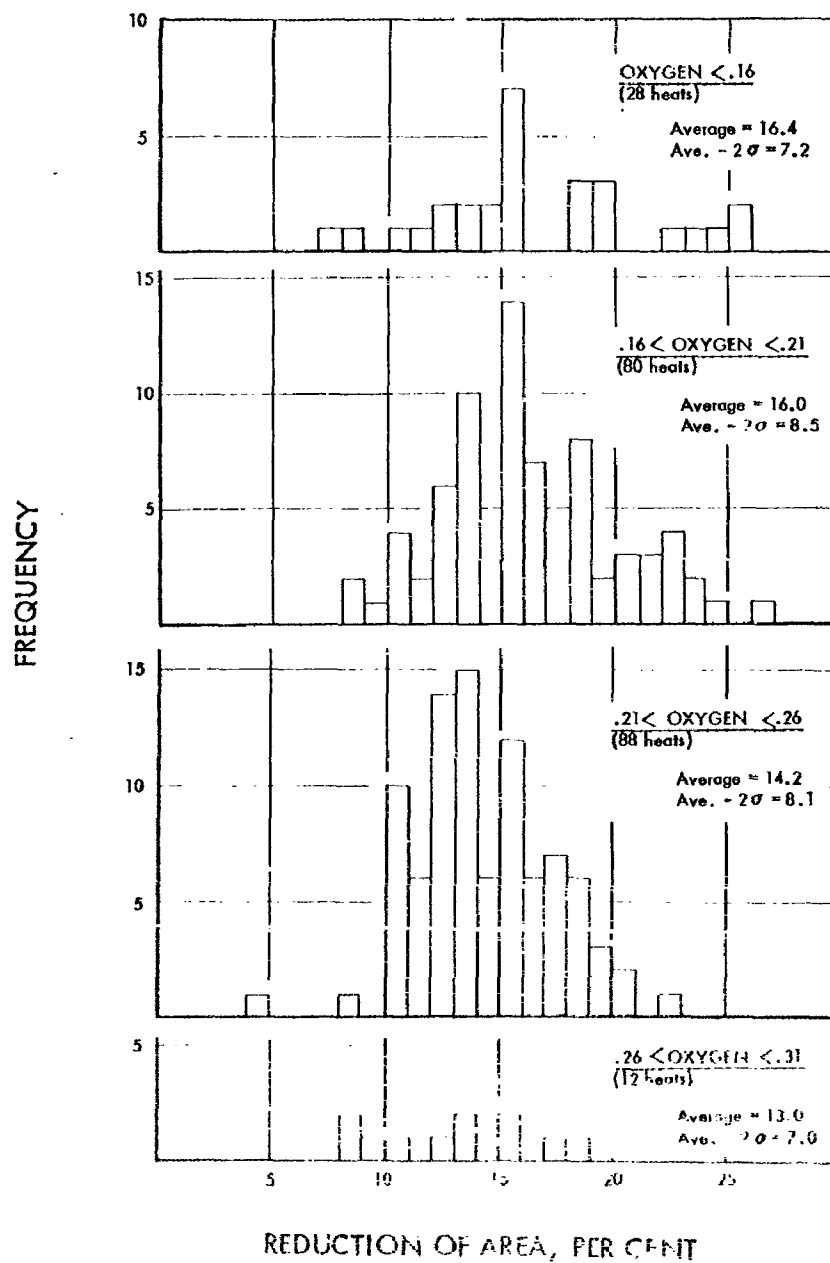
FREQUENCY DISTRIBUTIONS FOR REDUCTION
OF AREA FOR VARIOUS OXYGEN RANGES

FIGURE C12

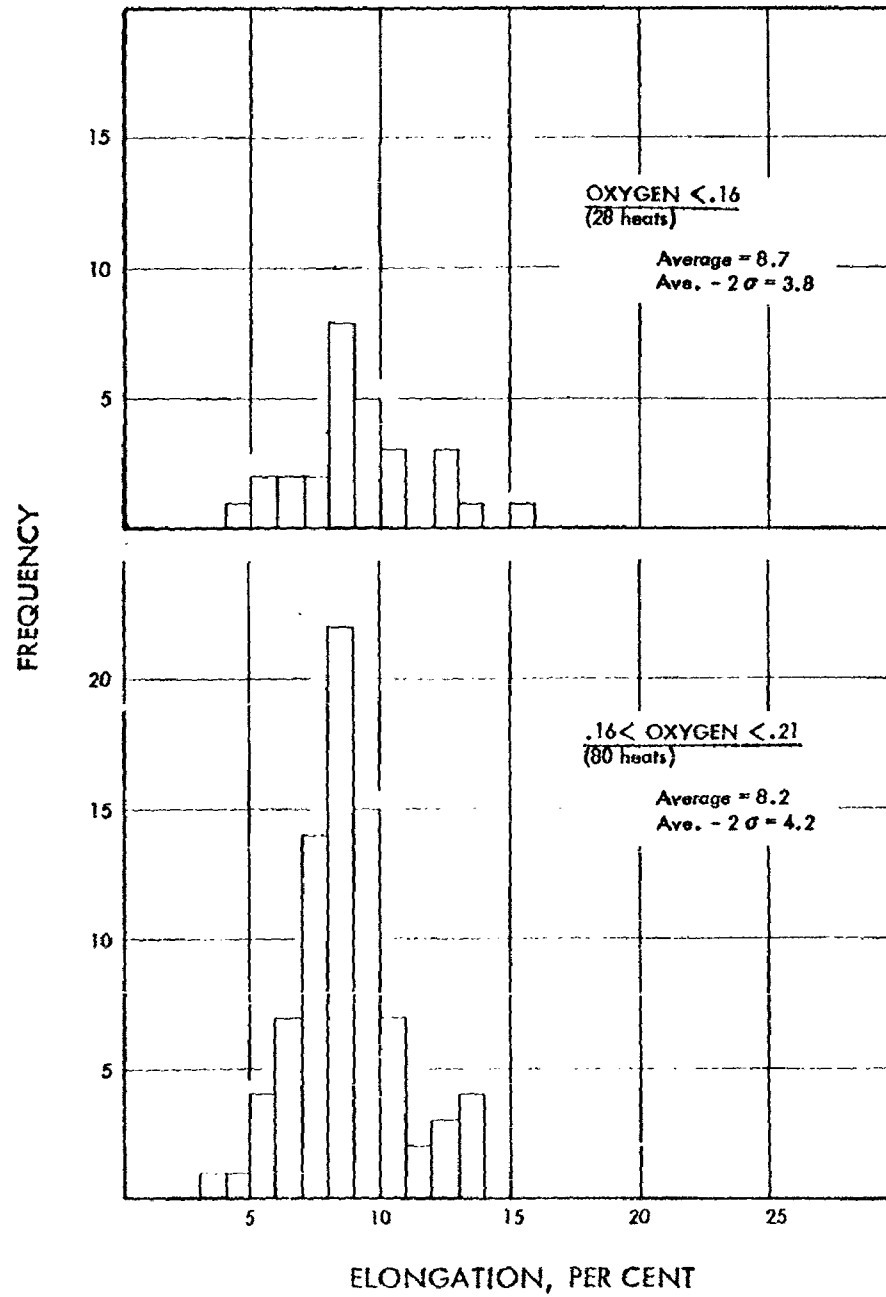
FREQUENCY DISTRIBUTIONS FOR ELONGATION
FOR VARIOUS OXYGEN RANGES

FIGURE C13

(FIGURE 12 CONTINUED)

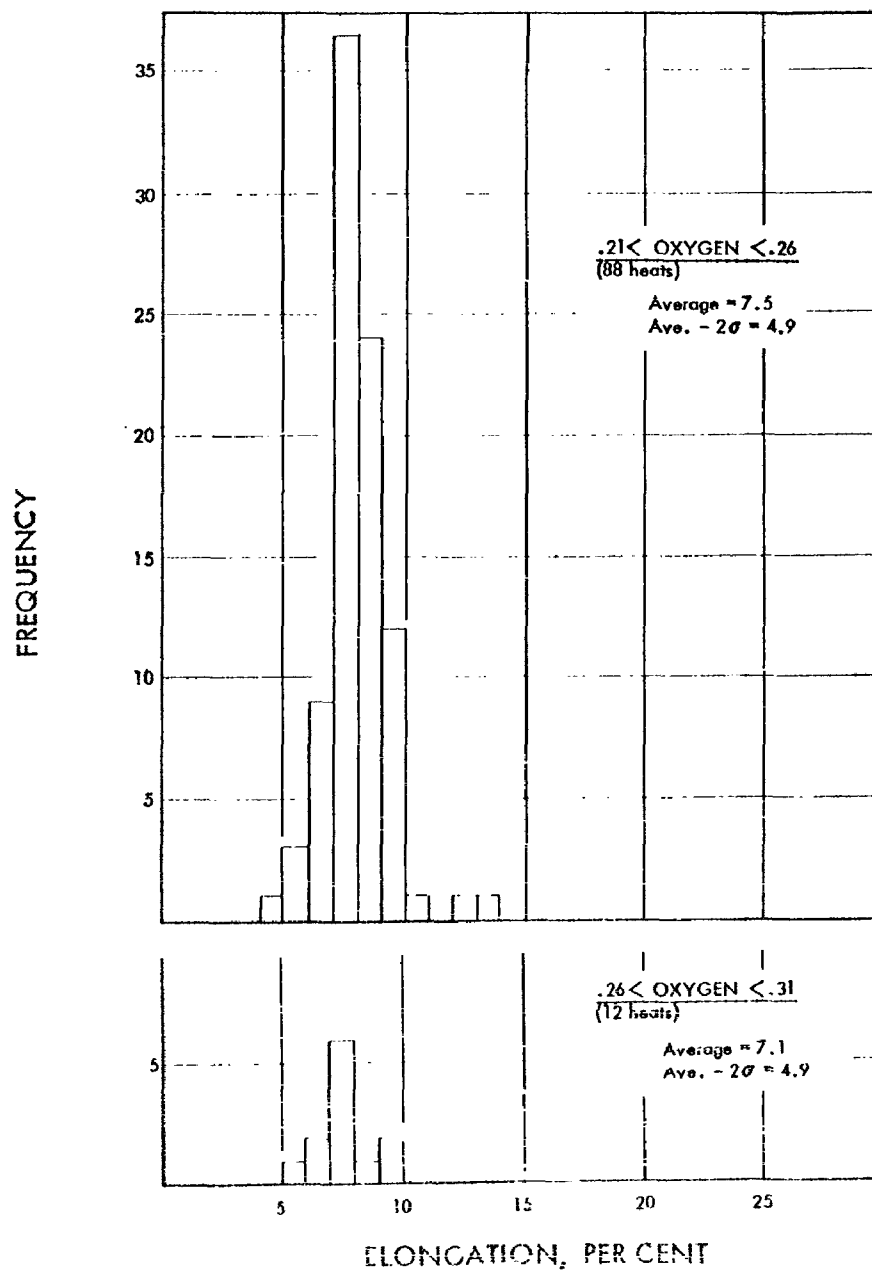


FIGURE C14

SCATTER DIAGRAM - NITROGEN VS. ULTIMATE STRENGTH

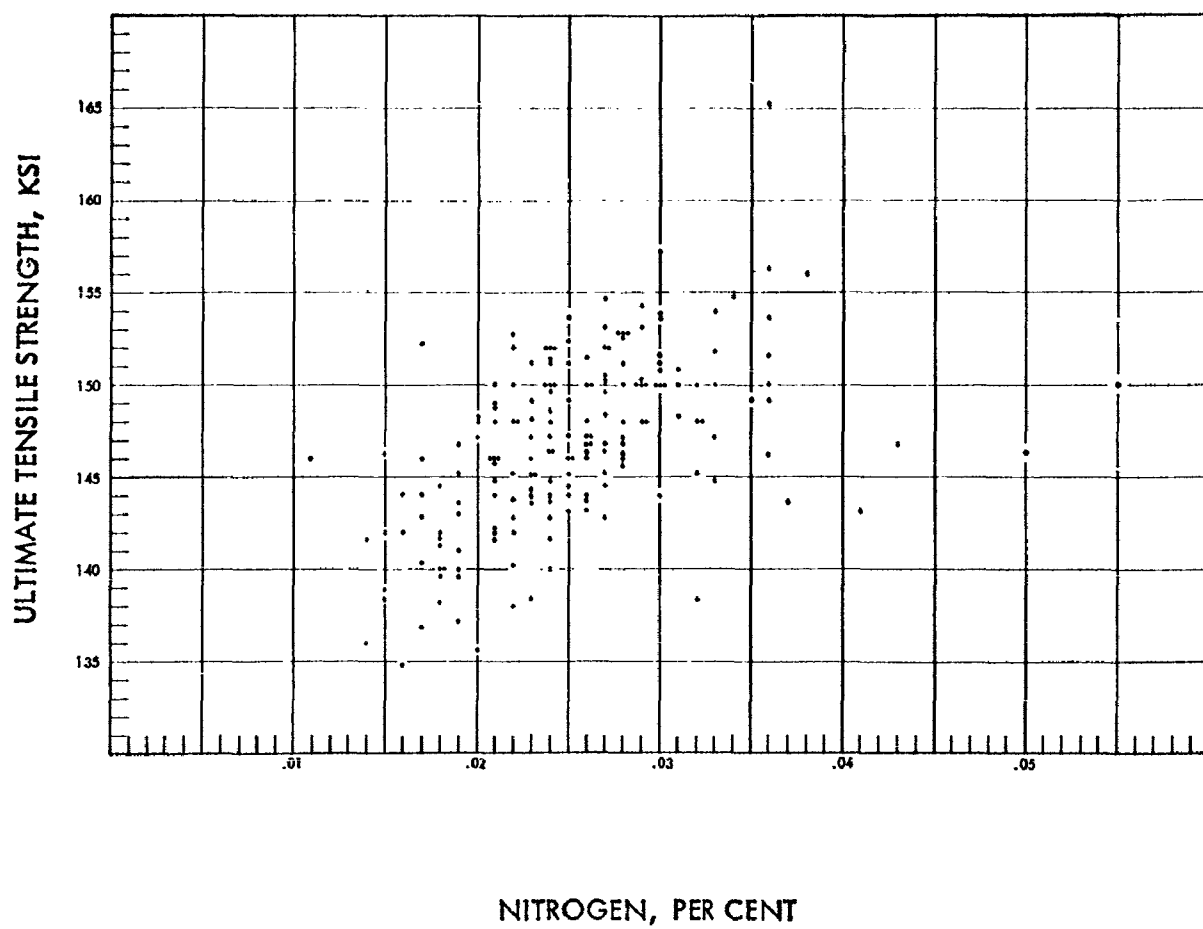


FIGURE C15

SCATTER DIAGRAM - NITROGEN VS. REDUCTION OF AREA

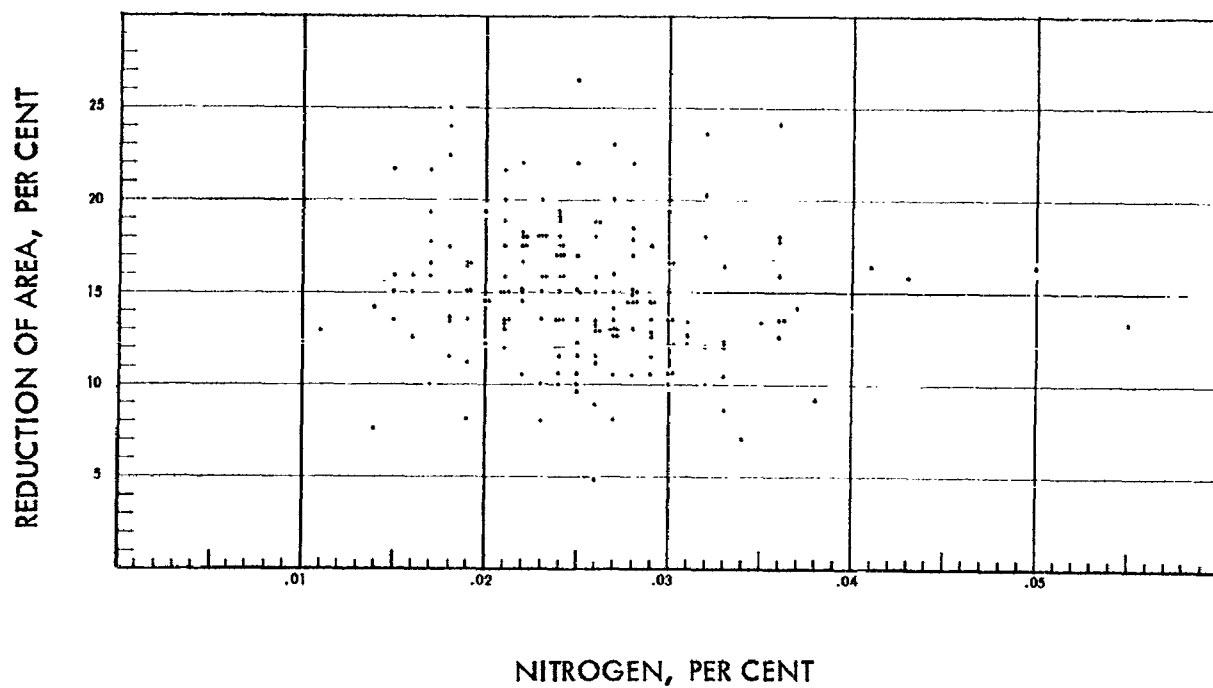


FIGURE C16

SCATTER DIAGRAM - NITROGEN VS. OXYGEN

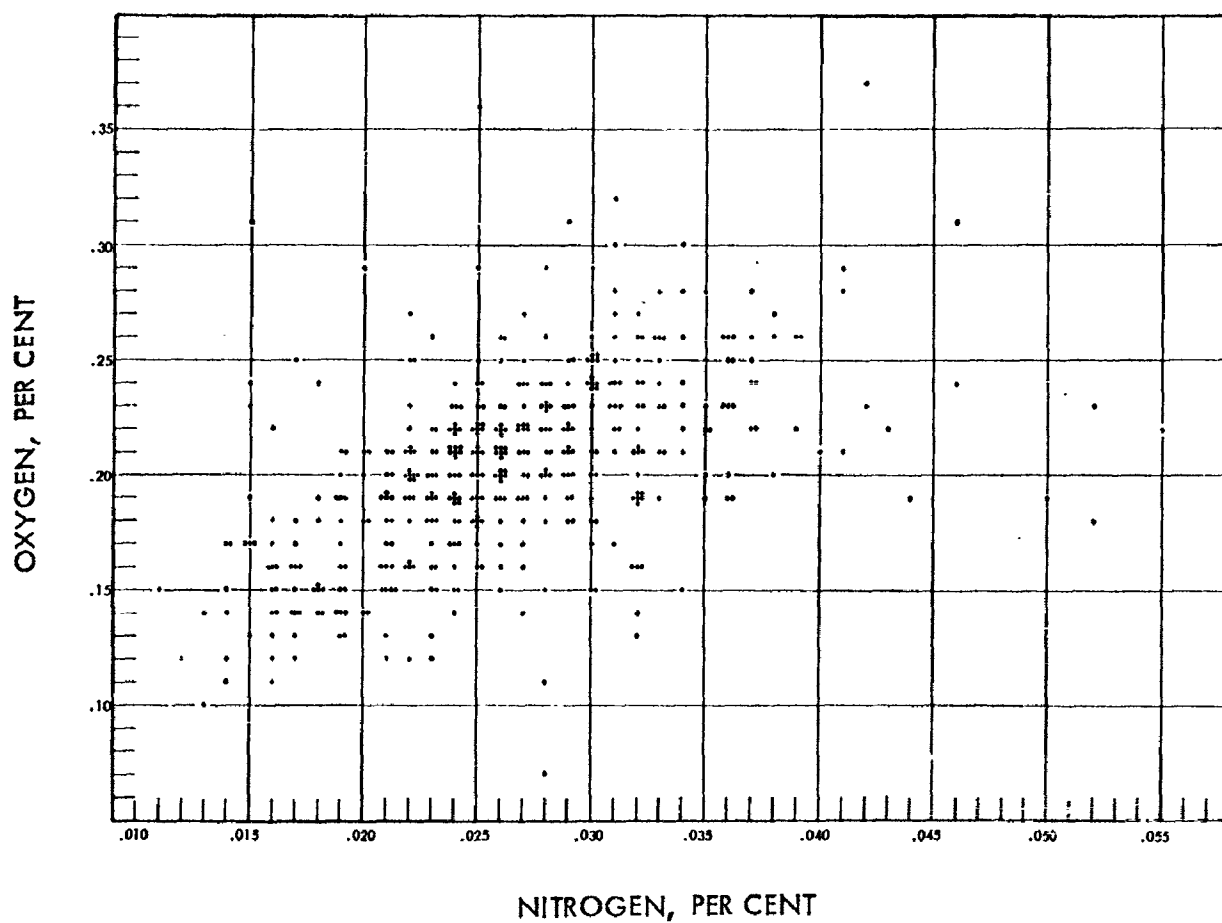
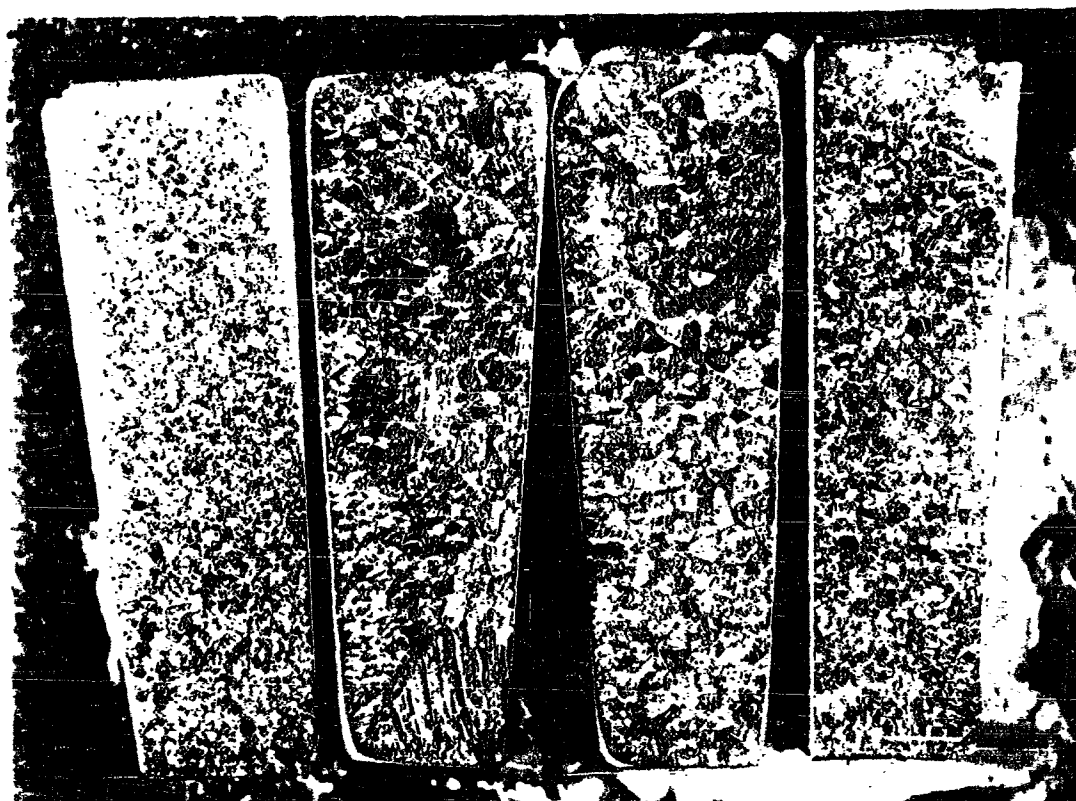


FIGURE C17
EFFECT OF COOLING RATE ON MACROSTRUCTURE



Copper
Mold

Rammed
Graphite
Mold

Shell
Mold

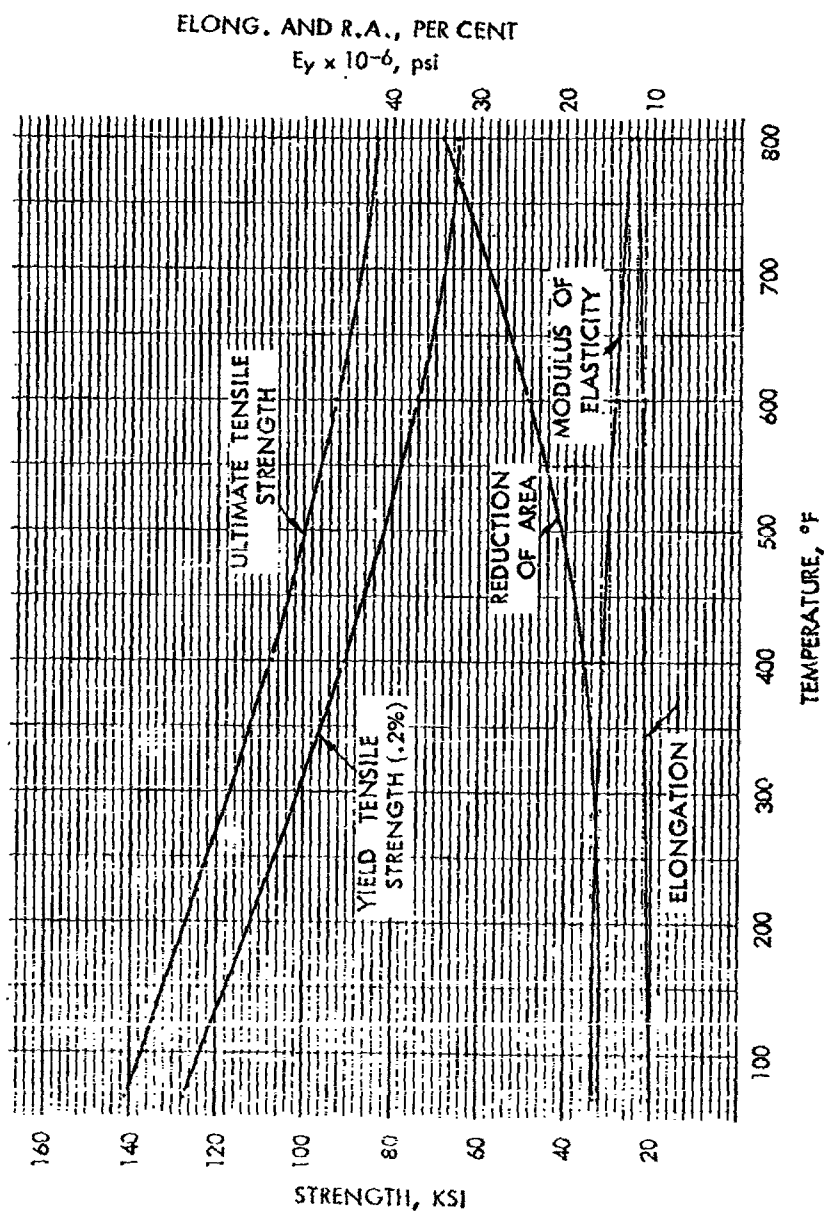
Machined
Graphite
Mold

Samples: Test coupons of C-4 alloy cast in different mold materials.

Mag. 1 X

Structure: Note refinement of grain resulting from fast cooling in copper mold and directional solidification developed in slower cooling of rammed graphite material.

FIGURE C18
ELEVATED TEMPERATURE PROPERTIES OF AS-CAST Ti-6Al-4V



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SECTION D

CASTING OF SHAPES

Selection of Components

Five parts were selected for use as developmental shapes in this program. The parts represent a variety of casting problems and are considered typical of structural members which would be desirable as castings.

The following parts were selected:

(1) Developmental Bracket

A sketch and a reduced size drawing of the Developmental Bracket are shown as Figures D1 and D2. This part is not an actual production component, but was designed specifically for development work on castings and forgings. As a shape, it is difficult to forge or cast because of the adjacent heavy and thin sections, the high thin flanges, and the thin confined web. The part is somewhat typical of airframe structural detail and is small and economical to produce, machine into a finished part, and test. The Developmental Bracket was used for much of the basic development work in this contract. As a titanium casting, the machined shape weighs approximately 2.2 pounds.

(2) Flap Track Support Link

A sketch and a drawing of this part are shown as Figures D18 and D19. This is a redesigned KC-135 Tanker-Transport structural brace, presently forged from AISI 4140 or 6740 steel and heat treated to the 180 - 200 ksi ultimate tensile strength range. Four of this configuration are used per airplane. The machined component weighs 2.9 pounds as a titanium casting.

(3) Sway Brace Case

This is a B-52 Bomber component, essentially a spring casing, presently used as an AISI type 410 steel rough centrifugal casting heat treated to 150 - 170 ksi ultimate tensile strength. The machined titanium part weighs 7.75 pounds. There are eight used per airplane. Conversion to a titanium casting would save 46 pounds per airplane. A sketch and a drawing of this part are shown as Figures D32 and D33.

(4) Elevator Torsion Fitting

A sketch and a drawing of this component are shown as Figures D36 and D37. The part is a redesigned structural component that converts rotation of an input shaft to change of attack angle of the horizontal stabilizer on the Bomarc missile. The fitting is designed for rigidity and is presently an AISI type 410 steel casting heat treated to 180 - 210 ksi ultimate tensile strength. One left and one right hand versions of this component are used per missile and weight 4.48 pounds each as machined titanium castings.

(5) Inboard Horizontal Stabilizer Rib

A sketch of this component is shown as Figure D40. The corresponding production component controls angle of attack and attaches the horizontal stabilizer to the fuselage of the North American Aviation A3J-1 Airplane. The production part was initially designed and produced as a titanium alloy forging heat treated to 150 ksi ultimate strength but was later changed to a steel forging to increase rigidity. The titanium forging weighed 160 pounds and was machined to 37 pounds. This component was selected as a large casting which approached the size limitation of the available casting equipment. The part was not satisfactorily developed as a casting.

Casting of The Developmental Bracket

A sketch and drawing of the Developmental Bracket are shown as Figures D1 and D2.

The Bracket was first statically cast in both machined graphite and rammed graphite molds using the Ti-6Al-4V alloy. These first castings were produced without the optional holes in the web, and contained dispersed shrinkage in the web. This design, when produced in a machined graphite mold with a machined graphite core to form the web and inside surfaces of the flanges, also had extensive cracks in the web area because the high strength of the machined graphite restrained contraction of the metal during cooling after solidification. The casting was again produced in the machined mold using a rammed graphite core with no cracking.

Following further tests to verify the dispersed shrinkage distribution in the web area, the patterns were modified to include the optional holes through the web and castings were then produced in both machined and rammed graphite molds. The addition of the holes greatly improved the feeding characteristics of the web and relieved the problem of web cracking when machined graphite cores were used.

Experience with this design made it apparent that the conditions which produced web soundness were borderline, since several of the castings produced had some shrinkage in the web area. A typical risering and gating technique for static casting the Bracket is shown in Figure D3. Risers were located at each of the lugs, which represent three isolated thermal centers. Gating to the mold was through the risers so that risers would fill last and thus provide a greater temperature gradient from casting to riser.

Casting made in machined graphite molds generally exhibited a surface roughness visually comparable to a 150 RMS finish. In general, the surfaces of castings produced in rammed graphite molds were rougher than those in the machined mold, in the order of 120 to 300 RMS.

FIGURE D1

DEVELOPMENTAL BRACKET

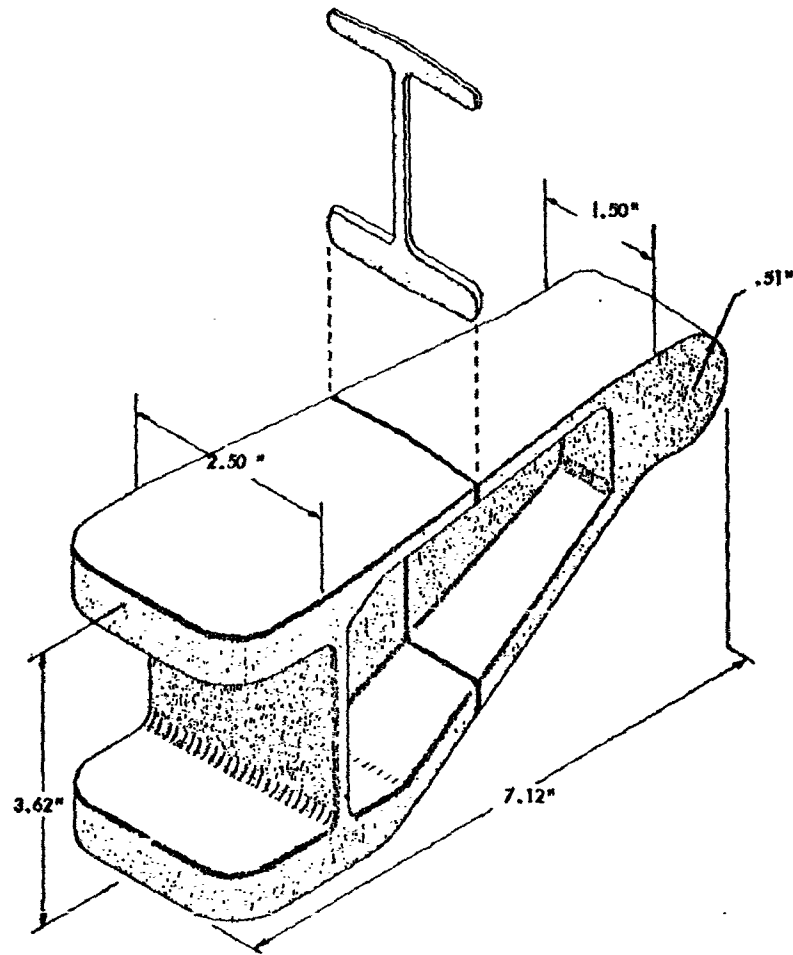
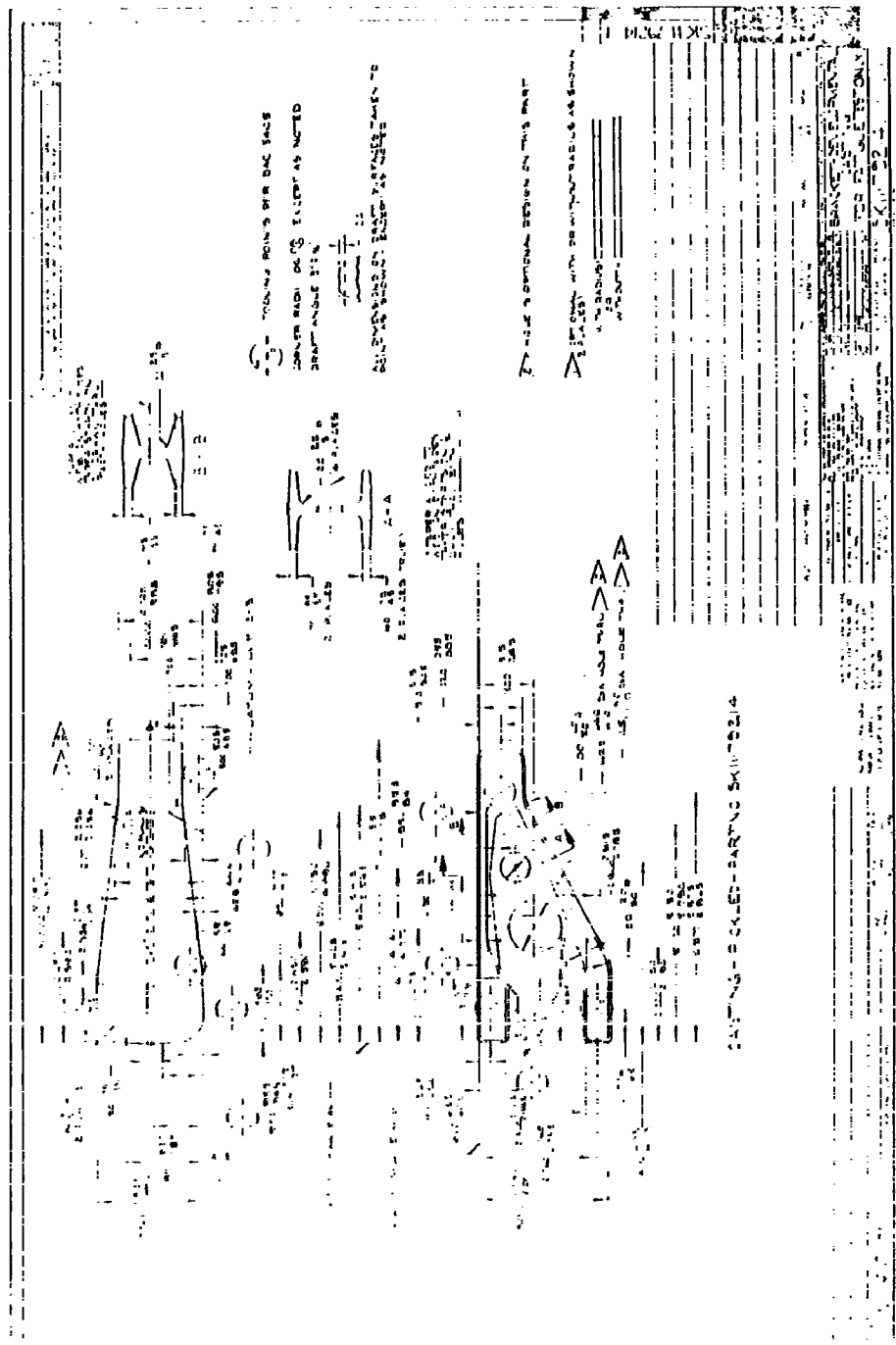


FIGURE D2

DEVELOPMENT BRACKET



Next, the shrinkage condition in the flanges was experimentally improved by adding a three degree taper to the exterior of the flanges, from the midpoint of the flange toward the riser connection. This reduced the shrinkage to very minor areas near the flange-to-web intersection. Incorporation of a one degree taper from the center of the web toward the flanges further improved the soundness to the point where shrinkage was no longer a problem in the web and flanges. The corebox was then modified to provide the flange taper on the inside of the flange to incorporate the web taper. It became increasingly obvious as shrinkage was eliminated that gas porosity was a major problem.

The initial trials of centrifuge casting the Developmental Bracket utilized machined graphite molds with machined or shell graphite cores to provide the web and inner flange surfaces. Centrifuge casting at 850 RPM (160 G's) resulted in nearly sound castings with slight shrinkage porosity in the heavy section at the small end of the part.

An additional casting was then made using a shell core with a larger ingate to feed the small end. This casting was also centrifuge cast at 850 RPM in a 16-inch diameter mold and had satisfactory soundness in the problem area. Surface quality of the casting was poor, because of penetration of metal into the core in the web and flange areas. The core material was easily removed from the casting by sand blast, but a rough surface remained on the casting. The roughness is attributed to lack of density at those surfaces of the core which are vertical in the corebox and did not receive adequate packing during shell molding.

A photograph of three casting made with the above mold arrangements is shown in Figure D4.

The Developmental Bracket was next cast by centrifuging in multiple cavity molds. The arrangement illustrated by Figure D5 was poured in a horizontal axis centrifuge casting furnace. The photograph shows the front of the mold as poured. The mold was rotated counterclockwise at approximately 900 RPM. This arrangement was used to provide "bottom pouring" into the mold cavity to prevent outward impingement of metal directly into the mold cavity. The metal enters the gates and is accelerated outward to the ends of the

gates, fills them, and then enters the mold with little turbulence. The castings produced were sound except for a shrinkage cavity in the large lug which is not connected to the ingate, caused by the necessity for this flange to be fed through the relatively thin sections between the two heavy lugs.

The castings shown in Figure D6 were cast in the vertical axis furnace and were gated and risered at each heavy lug. These castings were x-ray sound but had slight surface laps or shuts along the lower flange, caused by metal entering from the top gate and flowing to meet the metal from the bottom gates.

The casting setup shown in Figure D7 was a variation used in an attempt to prevent surface laps. However this setup produced heavy surface laps because of too rapid filling of the mold cavity. The angle of the surface lap indicated a necessity for repositioning the mold in addition to a new gating arrangement. The results from this casting experiment indicated that in centrifugal casting, it is necessary that each end of the mold cavity be at an equal radius from the central sprue to permit even filling of the mold cavity.

The casting setup shown in Figure D8 was to determine the effect of higher casting forces on the outside mold and a new gating arrangement on the inside molds. The as-cast surface from the outside mold was improved slightly, and that from the inside mold was greatly improved. It is apparent that a reverse gating system is necessary to introduce a low-turbulence flow of metal into the mold cavity.

The setup shown in Figure D9 did not improve the casting surface to any great extent. The single gating arrangement did not reduce the turbulence for proper filling of the mold cavity.

The setup shown in Figure D10 illustrates improvements made in gating systems and mold position as a result of previous experimental heats. The mold position was changed such that a line through the mold cavity would be perpendicular to a line from the center sprue. This position permits low-turbulence filling of the mold cavity. The gate pipe feeding the ingates to the mold cavity has been reduced in size after the first gate opening to create a slight pressure head at the first

ingate. This method of gating was to provide equal flow of metal into each of the ingates. However, the small ingates (1" O.D.) did not permit the metal to enter at the proper rate and some surface laps appeared as shown in the sketch in Figure D10.

The photograph in Figure D11 illustrates a setup similar to that shown in Figure D10 with a gating change that increased the ingate into the mold to 1 1/4" O.D. This increase in gating size further improved the surface finish obtained.

The casting setup shown in Figure D12 is the same as shown in Figure D11. The variation in casting was to increase the centrifugal forces by rotating at 215 RPM (25 G's force at the cavity). The surface finish of the castings produced was not improved over previous castings cast at 140 RPM (12 G's). The noticeable difference was that the surface laps were only on the straight flange and the angular flange. There was no change in metal penetration of the shell core or the rammed graphite mold. It was concluded that the relatively high rotating speed was not advantageous as there was no improvement in surface finish. The effects of centrifugally casting at high speeds will become important where molds are stacked inwards towards the center of the sprue, enabling rotation at high enough speeds to obtain x-ray sound castings at the inner locations without detriment to the molds rotating at high speeds nearer the periphery of the plate.

The photographs shown in Figure D13 illustrate the casting setup used to test a higher production experimental pour. The top photograph illustrates the arrangement of four molds on the central sprue and the bottom photograph shows the gating system, developed in the previous experimental casting setups.

Of the entire quantity of castings produced in these trials, only two experimental heats failed to yield x-ray sound castings. One of the two heats was rejectable because of gas porosity. It was centrifugally cast at only 5 G's force which is evidently too slow to completely remove gas porosity. The second heat was rejectable because of shrinkage porosity and was made with the molds reversed such that the risers had insufficient effect in feeding shrinkage prone areas.

Next, the Bracket was experimentally cast using a variety of mold setups aimed at the development of an improved gating system and of multiple-mold centrifugal production setups. Radiographically sound castings were consistently produced.

Figure D14 illustrates a setup of two castings produced with a two-ingate mold-filling system. The molds are attached to the periphery of the runner bars with vertical ingates feeding the risers. The runner bar is reduced in size after the first ingate, to create a pressure head and increase the metal flow through the first ingate, thus providing a more uniform filling of the mold cavity. This change in the gating system for the Bracket was made to prevent surface laps in the castings caused by turbulent mold filling. Examination of the castings showed them to be free of such laps.

Figure D15 shows a four-casting arrangement and an eight-casting arrangement used in developing multiple-mold production setups. Visual examination showed these castings also to be free of surface laps.

In this production-setup work, the approach taken was to use a casting method which has been proven successful for small pours (as illustrated in Figure D13), and adapt the method for increased numbers of molds until a maximum is reached as dictated by equipment size. Figure D16 illustrates the pouring setup used for casting a 16-mold production arrangement. This was found to be the maximum number of castings that can be made under the existing casting conditions. The setup consists of eight stacks of two molds each. The molds are attached to a runner bar and gate mold of rammed graphite. The lower photograph in Figure D16 shows the setup after pouring was completed. Very little metal was lost through splashing and spilling from the sprue.

Figure D17 is a photograph of castings produced in the setup shown in Figure D16. Visual examination showed these castings to be free of surface laps.

X-rays of the 16 castings made in the first maximum production pour revealed a void in 15 of them. These voids were all located in the same portion of the casting. The cause of the void has been traced to the

FIGURE D 3

TYPICAL RISERING AND GATING TECHNIQUE

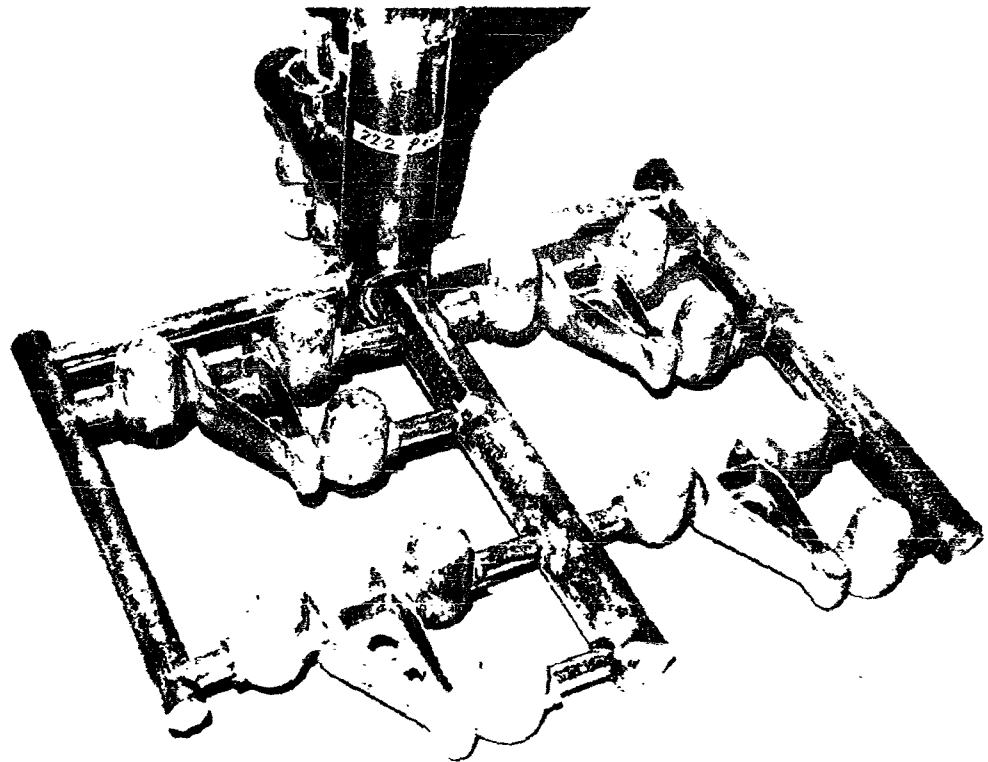
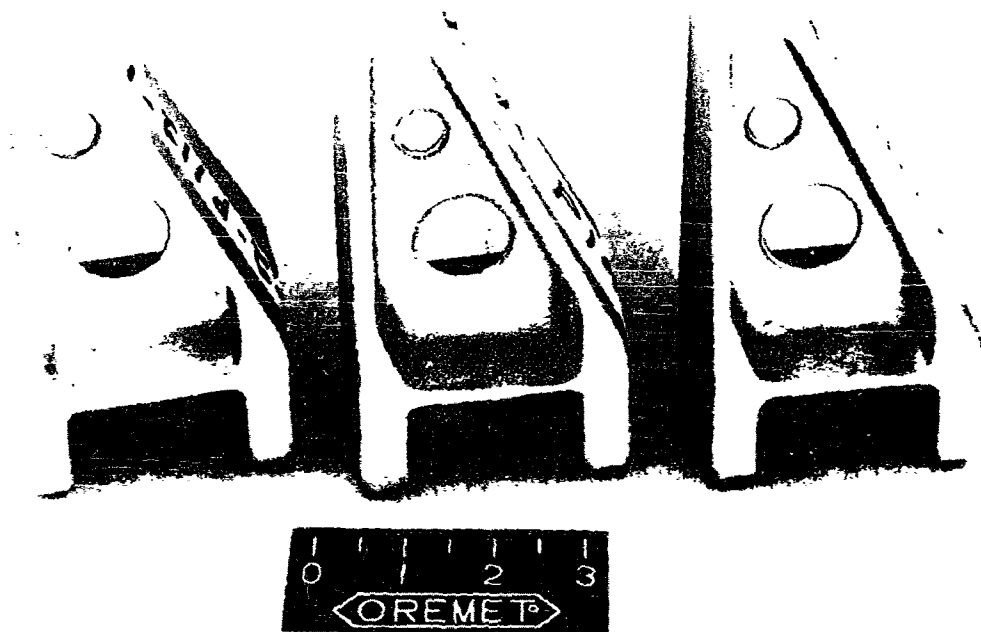


FIGURE D4

EXAMPLES OF SURFACE CONDITIONS
OBTAINED IN BRACKET CASTING TRIALS



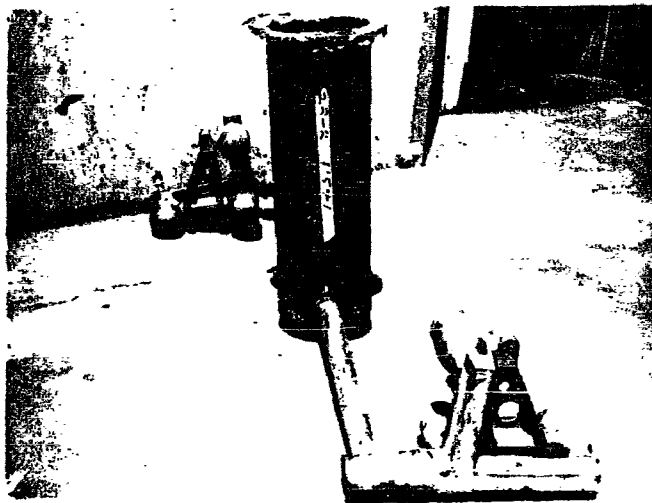
Left - Centrifuged, Machined Graphite Mold and Core
Center - Centrifuged, Machined Graphite Mold, Shell Core
Right - Static Cast, Rammed Graphite Mold and Core

FIGURE D5

CENTRIFUGALLY CAST DEVELOPMENTAL BRACKET



FIGURE D6



BRACKETS CENTRIFUGALLY CAST
AT 125 RPM (10G)

Rammed Graphite Mold, Shell Core

Below: Castings were X-ray Sound
Surface Showed Slight Laps

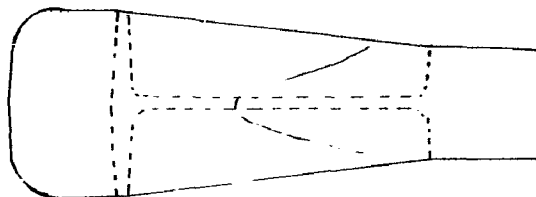
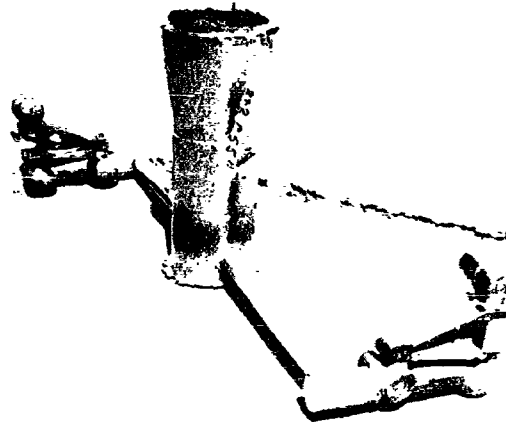


FIGURE D7



BRACKETS CENTRIFUGALLY CAST
AT 125 RPM (10G)

Rammed Graphite Mold, Shell Core

Below: Castings were X-ray Sound
Surface Showed Heavy Laps

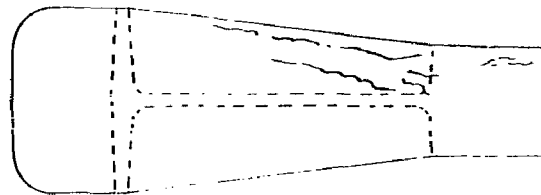
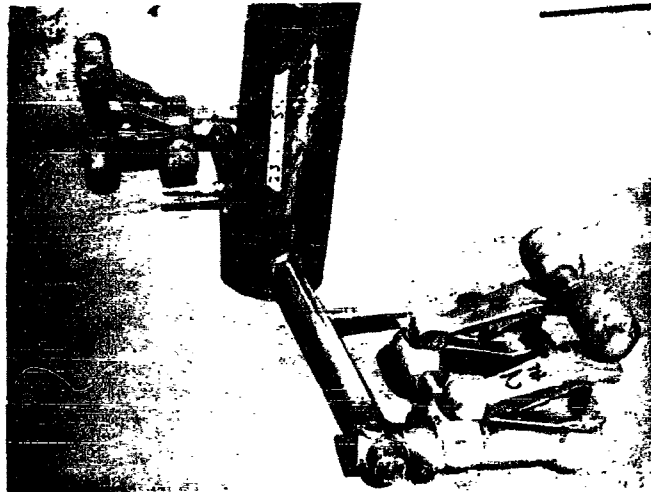


FIGURE D8



BRACKETS CENTRIFUGALLY CAST
AT 140 RPM (OUTER 126, INNER 106)

Rammed Graphite Mold, Shell Core

Below: Inner Castings Had Light Laps,
Outer Had Medium Laps. Both
were X-ray Sound

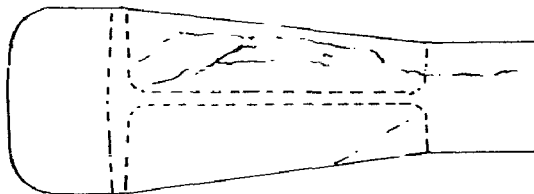
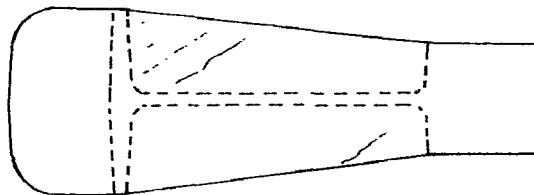
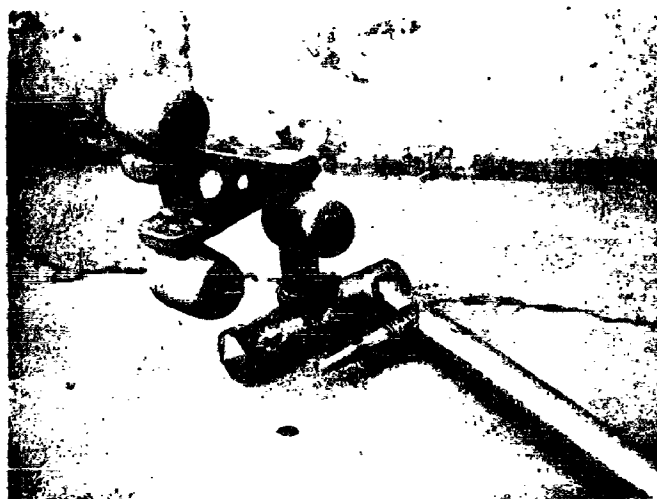


FIGURE D9



BRACKETS CENTRIFUGALLY CAST
AT 137 RPM (12G)

Rammed Graphite Mold, Shell Core

Below: Casting was X-ray Sound
Surface Had Medium Laps

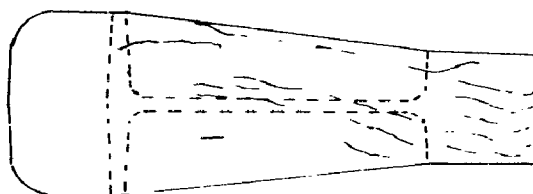
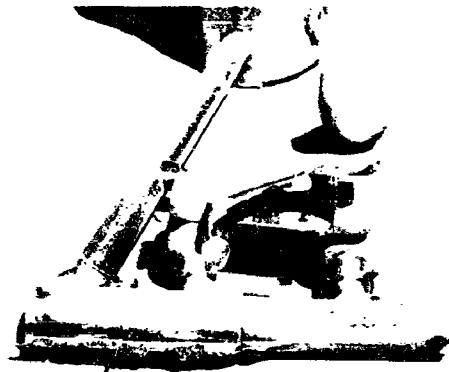


FIGURE D10



BRACKETS CENTRIFUGALLY CAST
AT 130 RPM (10G)

Rammed Graphite Mold, Shell Core

Below: Casting was X-ray Sound
Surface Had Slight Laps

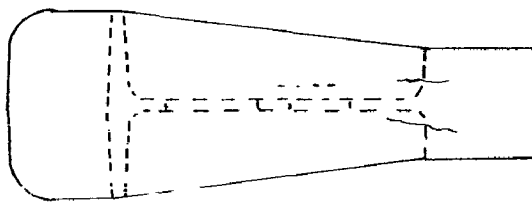
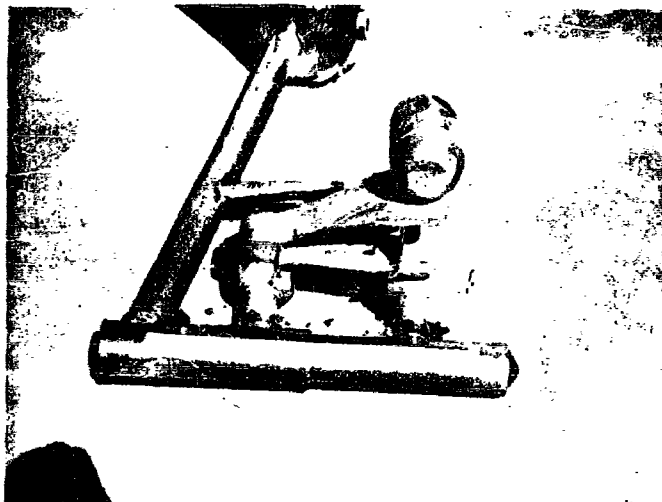


FIGURE D11



BRACKETS CENTRIFUGALLY CAST
AT 130 RPM (10G)

Rammed Graphite Mold, Shell Core

Below: Casting was X-ray Sound
Surface Had Slight Laps

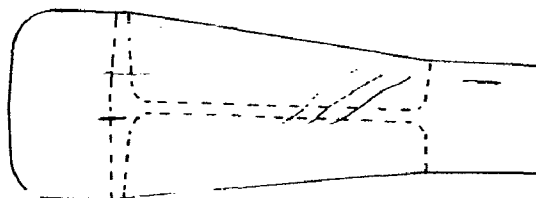


FIGURE D12



BRACKETS CENTRIFUGALLY CAST
AT 215 RPM (25G)

Rammed Graphite Mold, Shell Core

Below: Casting was X-ray Sound
Surface Had Medium Laps

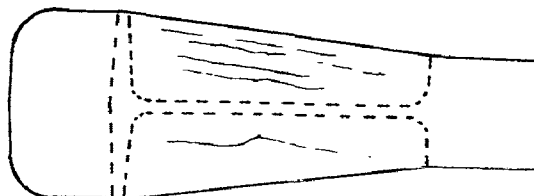
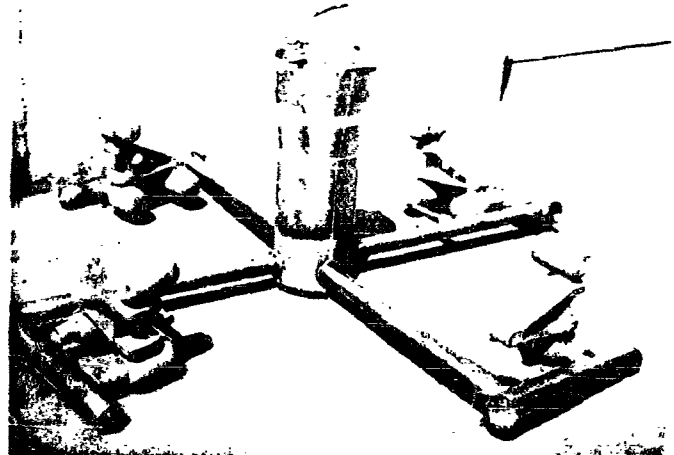


FIGURE D13



BRACKETS CENTRIFUGALLY CAST
AT 135 RPM (12G)

Rammed Graphite Molds, Shell Core

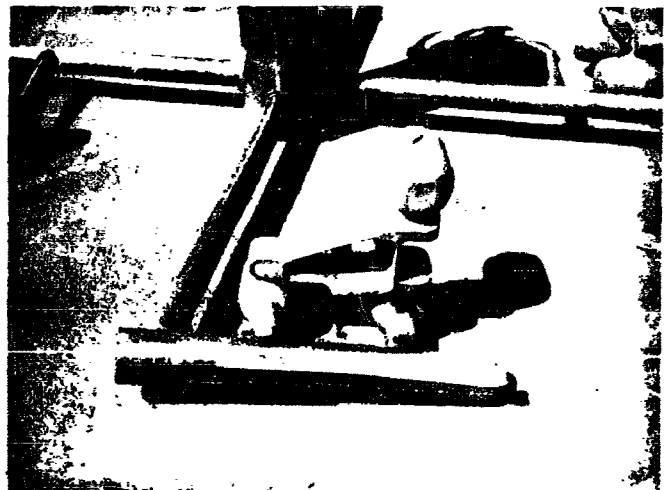


FIGURE D14

CENTRIFUGALLY CAST DEVELOPMENTAL BRACKETS

Two-Casting Arrangement

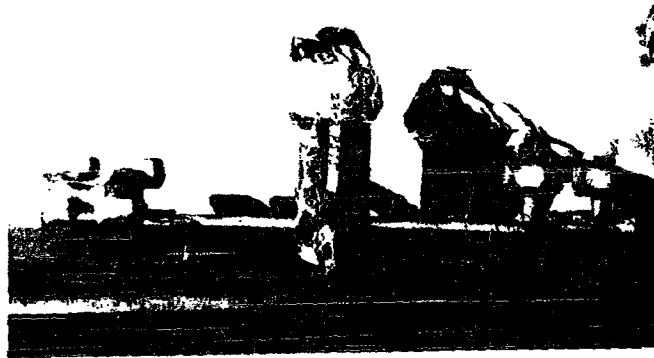
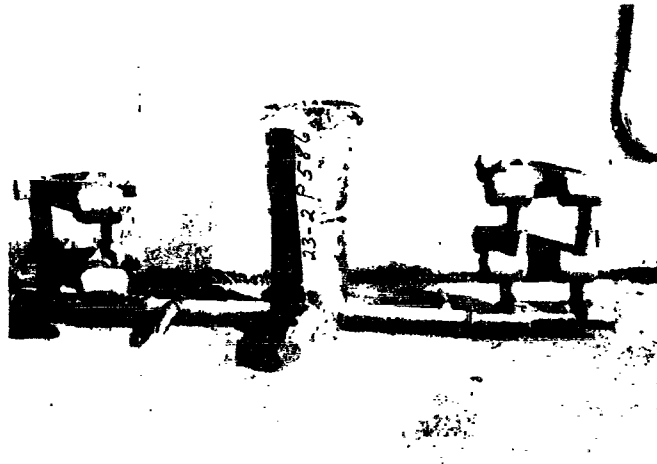
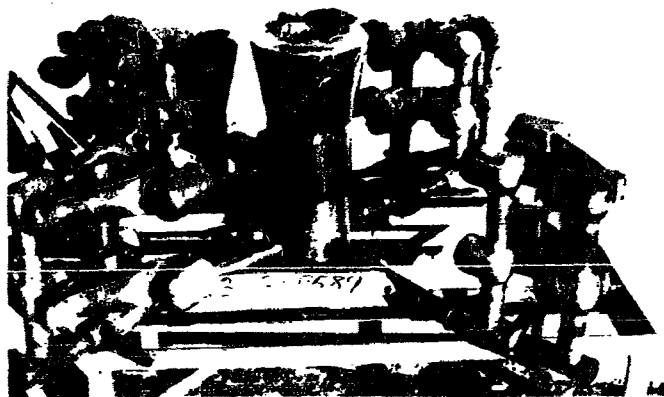


FIGURE D15

CENTRIFUGALLY CAST DEVELOPMENTAL BRACKETS



Four-Casting Arrangement



Eight-Casting Arrangement

FIGURE D16

SIXTEEN-MOLD CENTRIFUGAL SET UP
FOR DEVELOPMENTAL BRACKET



Before Pour

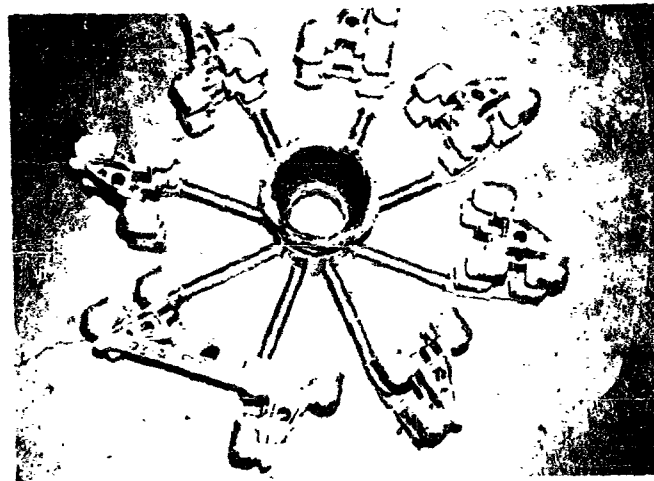


After Pour

FIGURE D17

CENTRIFUGALLY CAST DEVELOPMENTAL BRACKETS

Sixteen-Casting Arrangement



method of feeding the upper riser. An analysis of the gating system showed that a large volume of metal was being fed through a thin web to the riser, from where it was subsequently fed back to the casting. This situation caused improper thermal gradients to be set up across the riser and casting.

The gating system for this part was then revised to allow metal to flow to the mold cavity through each of the three risers as shown in Figure D6. The surfaces of castings made with this system were found to be free of visible laps, and x-rays showed them to be sound. This mold arrangement was used to produce the pilot production lot of 50 castings.

To obtain the production lot of 50 parts, it was necessary to cast five heats, a total of 80 castings. The rejection rate was approximately 33 per cent, of which half the rejections were because of surface defects and the remainder because of internal (radiographic) defects. The foundry yield (acceptable casting weight/weight of poured metal) was in the order of 14 per cent.

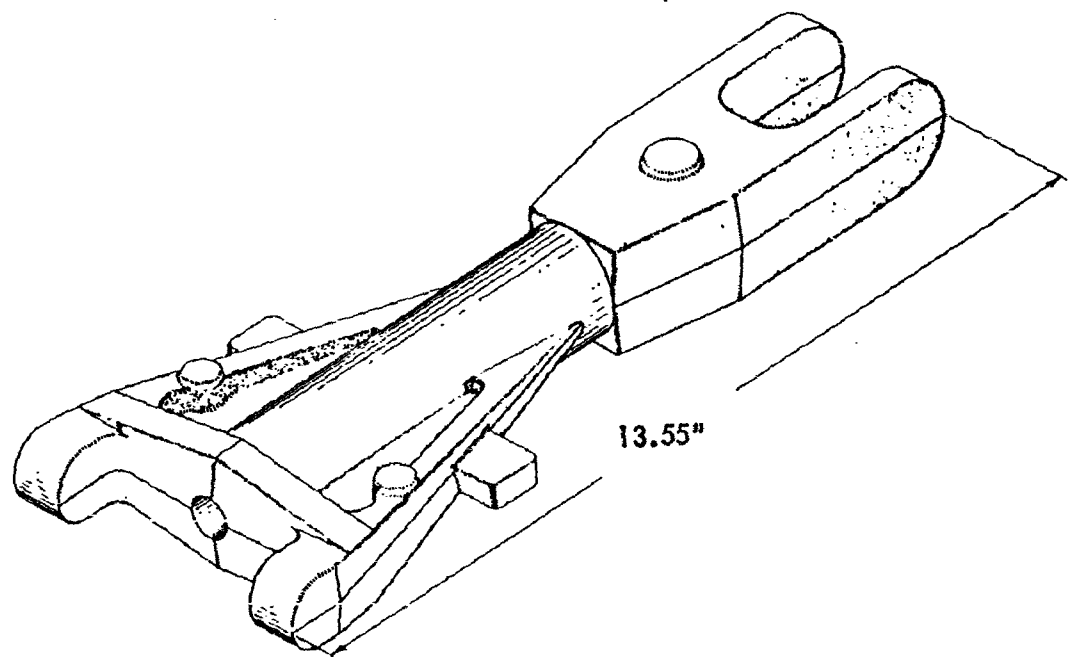
Of the castings shipped, none were entirely within the dimensional tolerances established as targets early in the program or within the tolerances later established as representative of this process, mostly because of pattern errors which were not corrected before beginning the production phase. All of the shipped castings were x-ray acceptable. The ranges of deviation from average of several specific dimensions on the 50 production phase parts was determined and is discussed in the section of this report dealing with development of specifications. Representative tolerances were statistically developed from these data.

Casting of The Flap Track Link

A sketch and drawing of this component are shown in Figures D18 and D19. Initial design of this casting included a cored hole through the cylindrical portion, but this requirement was removed in favor of machining the bore when it did not core satisfactorily.

FIGURE D18

FLAP TRACK SUPPORT LINK



The initial castings were made in rammed graphite molds with rammed and with machined graphite cores. The first results obtained were considered generally good. The tool hold-down lugs (along the flanges reinforcing the tubular section) represent isolated thermal centers that caused shrinkage voids. Other isolated shrinkage areas are located at the junction of the tubular portion and the transverse flange at the wide end of the part. A photograph of the first casting made is shown as Figure D20. The gating and risering setup is shown as Figure D21.

Next, a shell core was used in an attempt to cast the Flap Track Link to soundness. It had been established that the use of the shell core promoted soundness by reducing the chilling effect and extending feeding distance. The web areas at the large end of the link presented a borderline shrinkage problem with soundness being obtained part of the time. Some of the castings exhibited shrinkage at riser junctions, because of the difficulty of attaching properly sized risers to the casting without obscuring casting detail. Large heavy risers were added at the top of the casting at the heavy sections in an attempt to better feed the casting. The riser was necked down to retain casting detail and unfortunately the necked down portion of the riser closed off feed metal before the casting solidified.

The two tooling lugs extending from the sides of the casting were, at this time, problem areas in shrink porosity. Chilling of the tool lugs and use of shell cores promoted directional solidification to the point where the shrink void was located just inside of the casting at the riser junction. The location of the single shrink void in the middle of the tool lug is acceptable since these lugs are removed after the part has been machined, but since the location of the shrink void was partially in the casting when shell cores were used, an attempt was made to remove the shrink by supplementary gating from the riser located in close proximity to the tool lug.

Experimental castings were next made without coring. This method permitted the shrinkage to concentrate along the centerline areas where it is removed by machining the center of the tubular section. Castings of the link were made while varying the position of the mold. Most of the variations in positioning resulted in very poor surface finish due to cold lugs being formed by metal impinging on the sides of the mold cavity.

Figure D22A illustrates the shrinkage in the Flap Track Link cast in a rammed graphite mold with rammed core at the small end and a machined graphite core through the center bore. Six thermal areas are represented by the six localized shrink areas shown at the relatively isolated heavy sections. Slight dispersed shrinkage occurred in the thin triangular web area. The casting was side gated through four risers located as shown on the sketch.

The sketch of the Link in Figure D22B represents a variation to determine the effect of copper chills placed at the tool hold down lugs to remove the shrinkage areas present. The copper chills moved the shrinkage area into the casting and reduced the size of the voids slightly. The six localized shrinkage areas remained.

Figure D22C represents a Link cast in a machined graphite mold with 2-1/2-inch diameter risers attached. The only noticeable difference in casting in the machined graphite mold was the change in location of the shrinkage at the tip of the thin triangular web area into the body of the casting.

Figure D23A arrangement was cast without a core in the center section to determine the possibility of isolating shrinkage at the center of the casting. This Flap Track Link was cast in a rammed graphite mold preheated in a vacuum heating oven to 320°F. The major shrinkage areas occurred along the center line of the casting. The previous shrinkages at the tip of the triangular web shaped area and near the core at the wide end of the casting were removed. The two shrinkage areas at the tool hold-down lugs remained with some change in location toward the center of the casting. General dispersed shrinkage again occurred in the web area of the casting.

The Flap Track Link illustrated in Figure D23B had an extra riser attached to the top of the casting at the heavy section near the narrow end of the Link. The riser was necked down slightly at the point of junction with the casting to give a "Abolition Core" effect to permit easier removal. The old mold cavity fed into all the shrinkage in that portion of the casting. The riser was necked back into the riser. It is not possible to get rid of the shrinkage in the casting at that location without the use of a core. The shrinkage was moved to the extra riser. In this experiment, the shrinkage was moved to the extra riser and the large center section of the casting was free of shrinkage.

gradients conducive to removal of the shrink areas at the tip of the web area. Isolated shrink remained near the core at the wide end of the Link. The shrinkage areas in the tool hold down lugs also remained.

The illustration in Figure D23C is of a second experimental casting without the core through the center. Shrinkage was isolated along the center line with the exception of the tool hold down lugs and the web. Figure D24A is the same casting x-rayed after drilling through the center of the casting to remove shrinkage.

Figure D24B represents a modification of the Flap Track Link by varying the method of gating. In attempting to remove the shrinkage area in the tool lug, a slot from the riser to the tool lug was milled into a rammed mold to provide a feed channel and gate. The shrinkage area in one tool lug was absent and in the other lug it developed as an elongated center shrink in the head along the thin web section. One isolated shrinkage area still remained at the wide end of the flange. The shrinkage had been reduced to one side of the casting, possibly indicating incorrect flow of metal into the mold cavity and the setting up of higher thermal gradients on one side of the casting.

Figure D24C is an illustration of a core modification in an attempt to remove shrinkage by a larger taper towards each end of the casting where risers are located. A two piece core of shell graphite and machined graphite was used. The shrinkage was generally isolated to the tip of the triangular shaped web areas.

Figure D25A illustrates a variation in the core by machining a straight machined graphite core and using it in conjunction with a shell graphite core at the small end of the Link. The straight center core causes the shrinkage to be isolated at the tip of the thin triangular web section. The same 7/8-inch diameter center core with a slight modification as in Figure D25B eliminated shrinkage in the casting. The variation in Figure D25B reduced the concentration of metal by coring a portion of the critical area and establishing a favorable directional solidification condition. A general gas condition has prevented definite soundness conclusions to be reached.

The photograph shown in Figure D26 is of a static casting setup with gating and risering attached. Using the static casting method, the Flap Track Link can be produced with shrinkage only in areas which are subsequently machined out. The major problem in static casting, however, is the presence of dispersed gas porosity. The problem is an uncontrollable factor influencing the soundness quality of all of the statically cast parts. The sketch in Figure D26 illustrates areas where shrinkage is typical. Further experimental static casting was discontinued in favor of centrifuge casting to solve the gas porosity problem.

Next, experimental pours of this part were made in a vertical-axis centrifugal furnace. Figure D27 illustrates the setup of the part and the attached gating and risering. The mold was made using the same pattern and risering as is used for static casting. The gating arrangement was revised and repositioned to accommodate the centrifuge arrangement. A detrimental centerline shrinkage area occurs at the narrow end of the casting as shown in Figure D27. The method used to gate the casting was not satisfactory and medium surface laps occur along the surface of the part. A modification to introduce a smooth low-turbulence flow was necessary to remove the surface lap defect and produce a smooth cast surface.

Figure D28 illustrates a different experimental setup poured in the vertical-axis centrifugal furnace. The gating, risering, and mold position were modified in an attempt to arrive at a workable pouring setup. The sketch illustrates the location of resulting centerline shrinkage. The major shrinkage area occurred in an area where normal machining would not remove the defect. The risers located at the narrow end of the Link were intended to feed the area where shrinkage occurred and to feed the two flanges at that end. The thin section adjacent to the risers prevented adequate feeding of the heavier section where major shrinkage occurred. This gating system provided a low-turbulence method of pouring the casting and produced an excellent surface finish. The Link was free from porosity.

A small riser was next added to the Link of the centerline of the wide end to control shrinkage at that location. A comparison of Figures D29 and D30 will show the location of this riser and its effect on

shrinkage in the area. The position of the parts shown in Figures D29 and D30 during casting was such that the casting was self risering with the exception of the wide end, which was located nearest the center sprue. The shrinkage void at the riser, shown in Figure D30 could be removed entirely by enlargement of the riser.

Further modifications in the risering and gating system for this part are shown in Figure D31. In the system shown in the lower photograph of Figure D31, the large risers at the narrow end of the Flap Track Link were modified. This modification did not provide a sufficient volume of metal to feed the narrow end, allowing shrinkage to occur. To eliminate this shrinkage, small risers were added at the narrow end of the Flap Track Link. The risers at the narrow end of the part in the top photograph of Figure D31 are unnecessarily large and contribute little to the soundness of the casting.

The method of gating the Flap Track Link at the outer extremity of the casting, as shown in Figure D31, is necessary to achieve a bottom pour with low turbulence as metal enters the mold cavity. Gate sizes are varied to cause an even flow of metal into both of the ingates.

Next, a preliminary dimensional analysis of the Flap Track Link was made. The general trend was for the parts to be oversize, indicating that the combined mold and metal shrinkage averages slightly less than $3/8$ inch per foot.

The pattern equipment was modified prior to casting pilot production heats. Risers were added to the narrow end of the Link and a riser was added to the center of the wide end to reduce the large shrinkage void occurring at that point. A $1/8$ -inch-radius fillet was added to the edges of the pattern to provide a mold with rounded edges, thus preventing a mold crushing problem and possible graphite inclusions.

The production run set of Links was to be made in five heats of twelve castings per heat. Mishaps and rejections caused an increase to six heats. A total of 50 parts was produced. Of the 72 castings made, 50 were accepted. Seventy-five per cent of the

rejections were because of mold shift during casting and incomplete filling, the remainder were rejected because of grinding errors. The overall rejection rate was 35%.

The castings were centrifugally cast twelve molds per heat. The molds were rotated at 130 RPM to obtain 10 G's on the center of the part.

X-ray quality of the accepted castings was good. Dimensional accuracy was fair but not within the specified tolerance. Tolerances are further discussed in the section on specifications.

FIGURE D20

FLAP TRACK SUPPORT CAST IN RAMMED GRAPHITE MOLD



FIGURE D21

TYPICAL GATING AND RISERING
SETUP FOR FLAP TRACK SUPPORT LINK

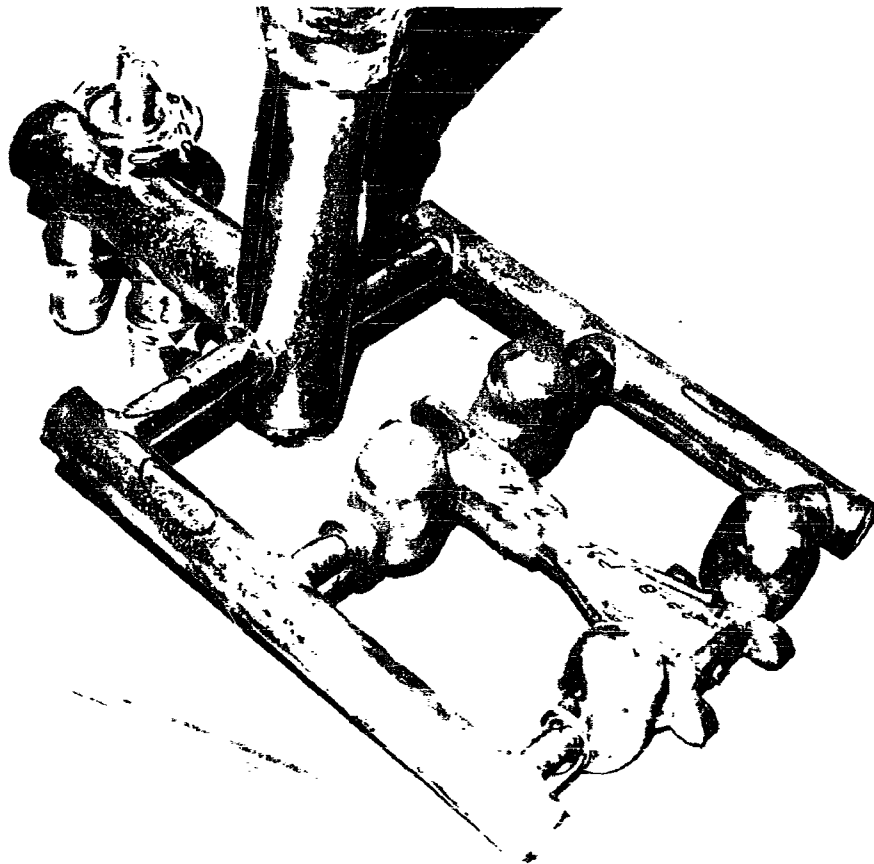


FIGURE D22

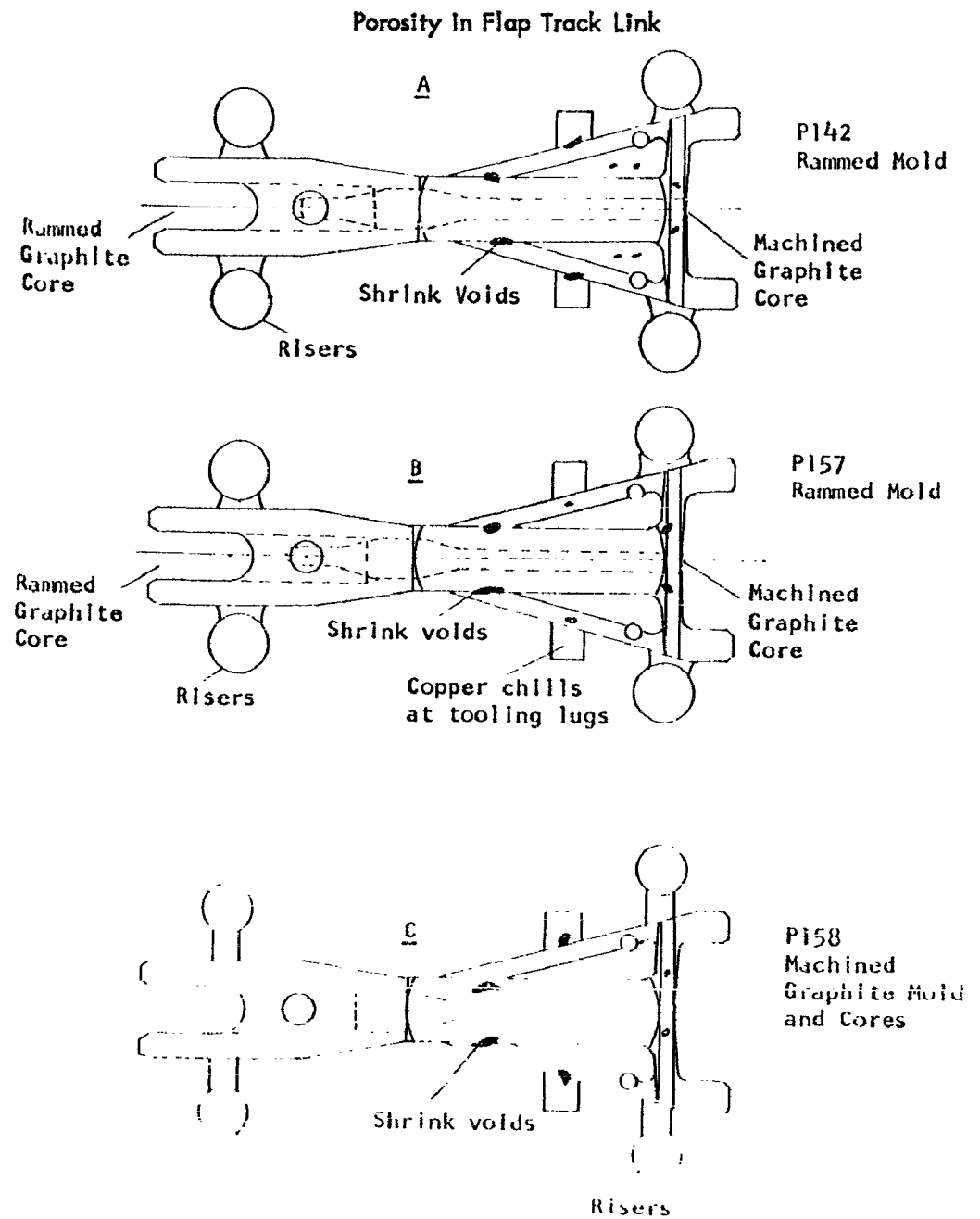


FIGURE D23

Porosity In Flap Track Link

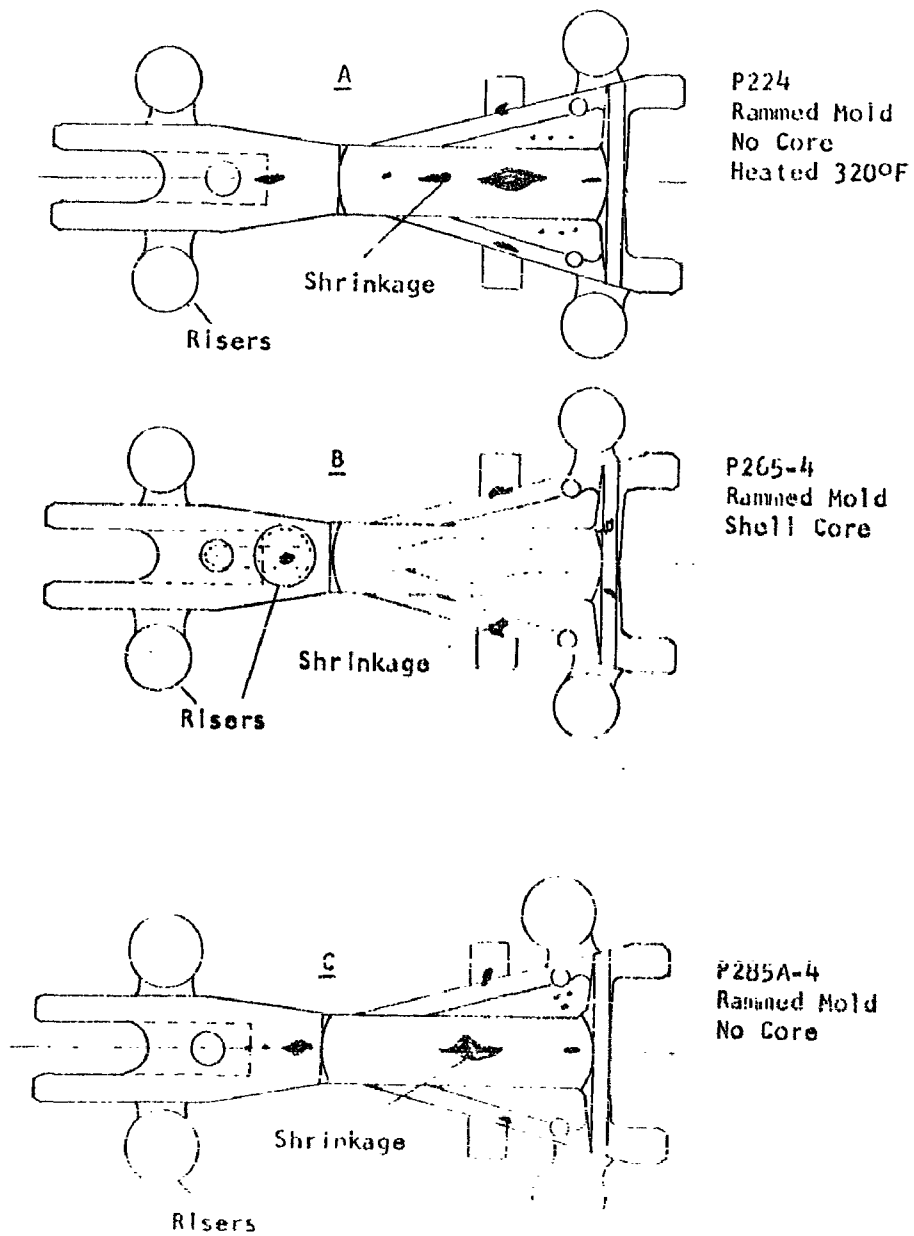


FIGURE D24

Porosity In Flap Track Link

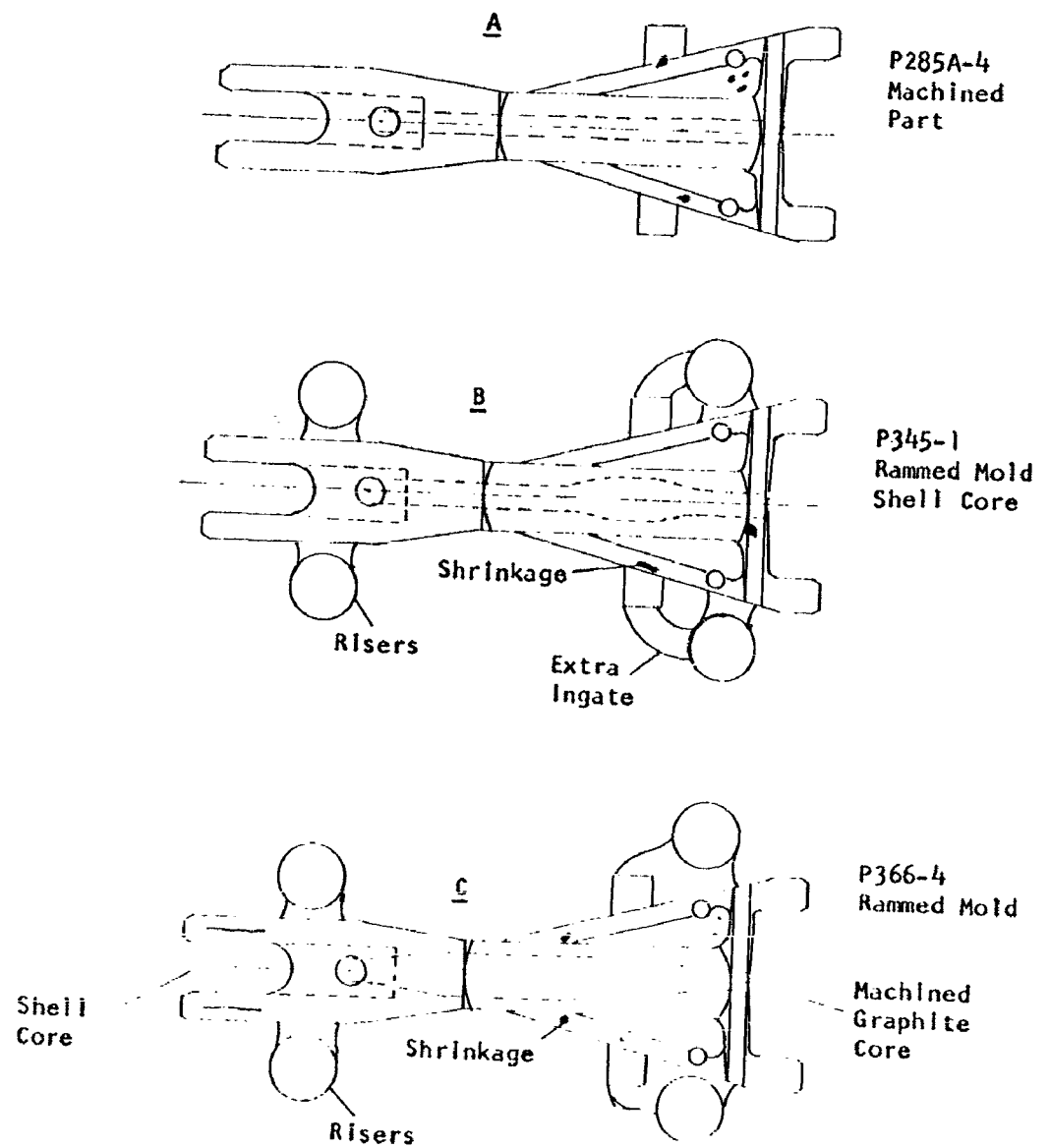


FIGURE D25

Porosity In Flap Track Link

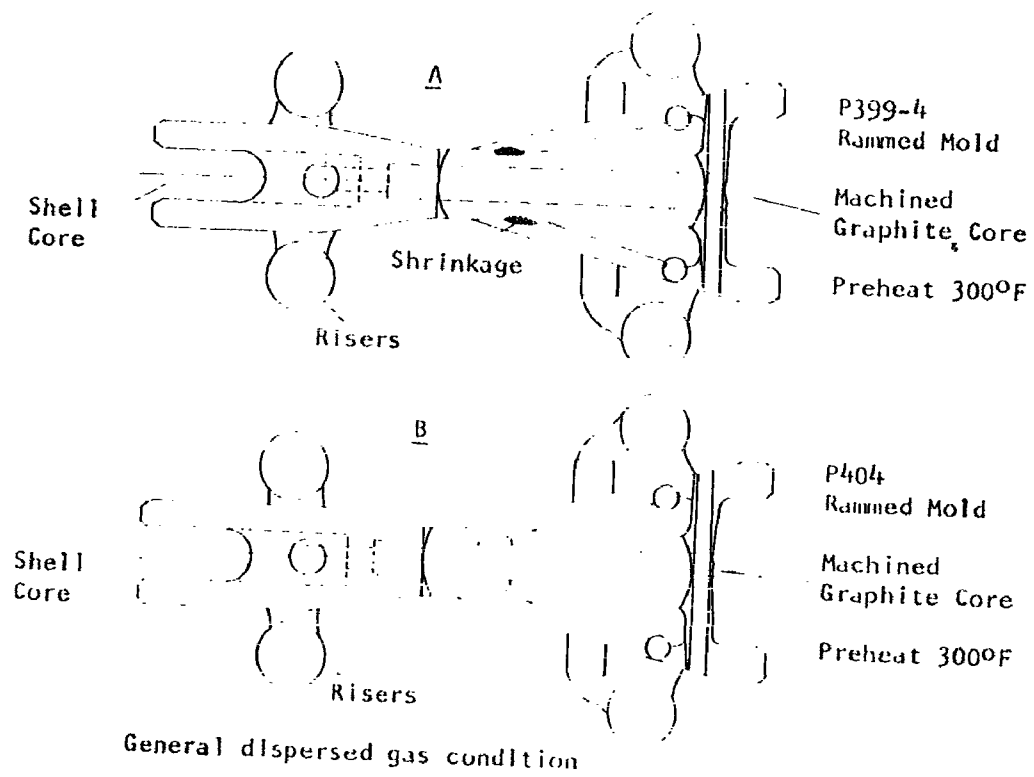


FIGURE D26



FLAP TRACK LINK, STATICALLY CAST

Rammed Graphite Mold, No Core

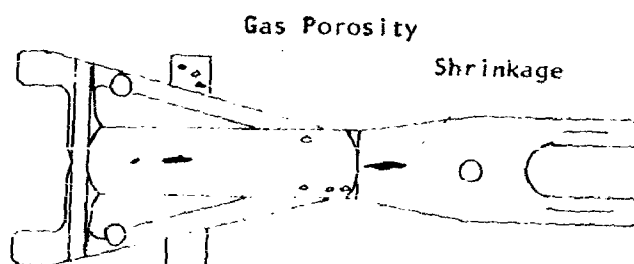


FIGURE D27



FLAP TRACK LINK CENTRIFUGALLY
CAST AT 150 RPM (10G)

Rammed Graphite Mold, No Core

Results Below

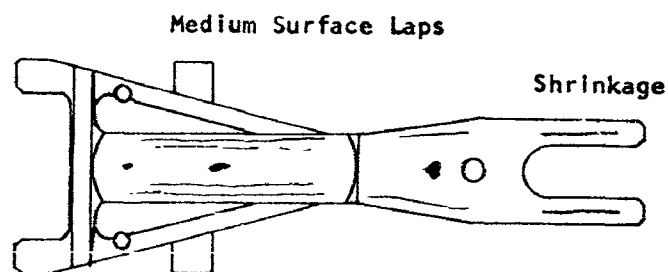


FIGURE D28



FLAP TRACK LINK CENTRIFUGALLY
CAST AT 110 RPM (10G)

Rammed Graphite Mold, No Core

Results below

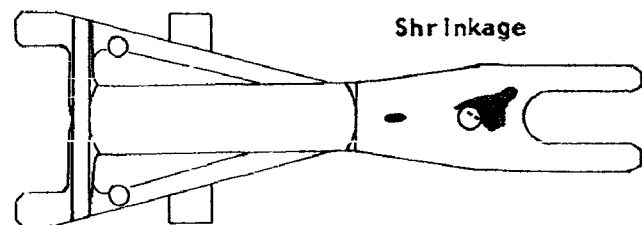


FIGURE D29

X-RAY OF FLAP TRACK LINK SHOWING
SHRINKAGE IN CYLINDRICAL SECTION

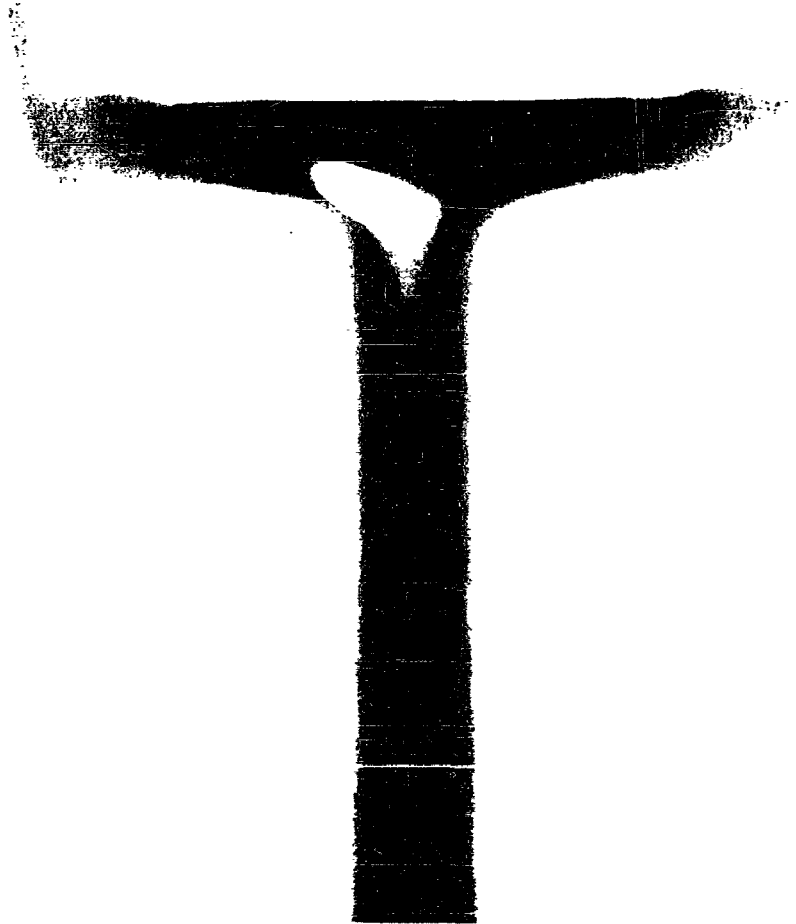


FIGURE D30

X-RAY OF FLAP TRACK LINK SHOWING
SHRINKAGE ADJACENT TO INADEQUATE RISER

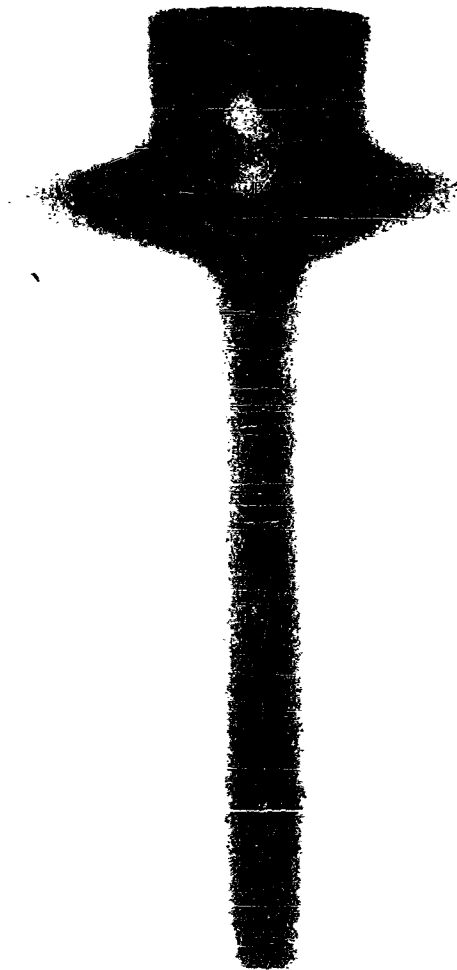
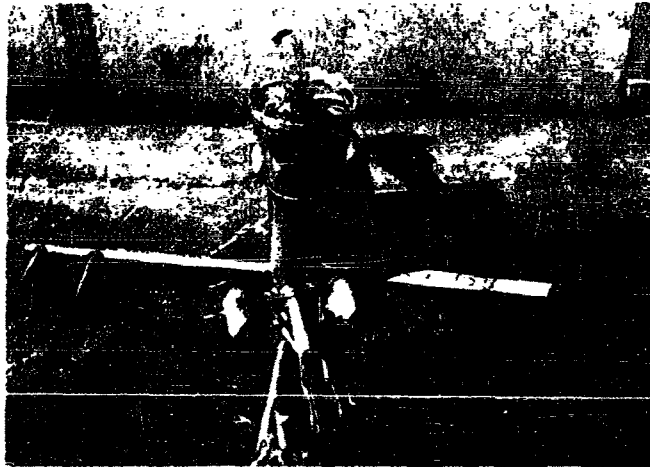
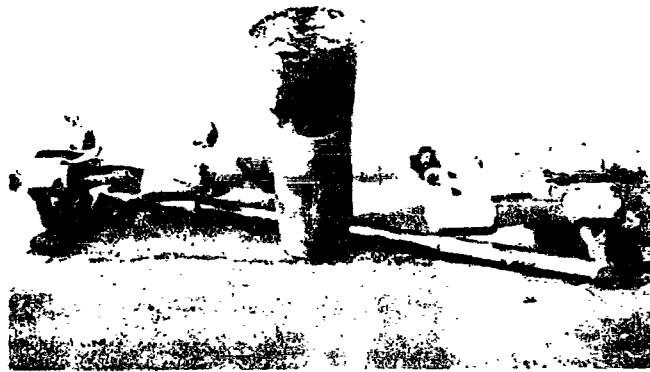


FIGURE D31

CENTRIFUGALLY CAST FLAP TRACK LINKS



Casting of The Sway Brace Case

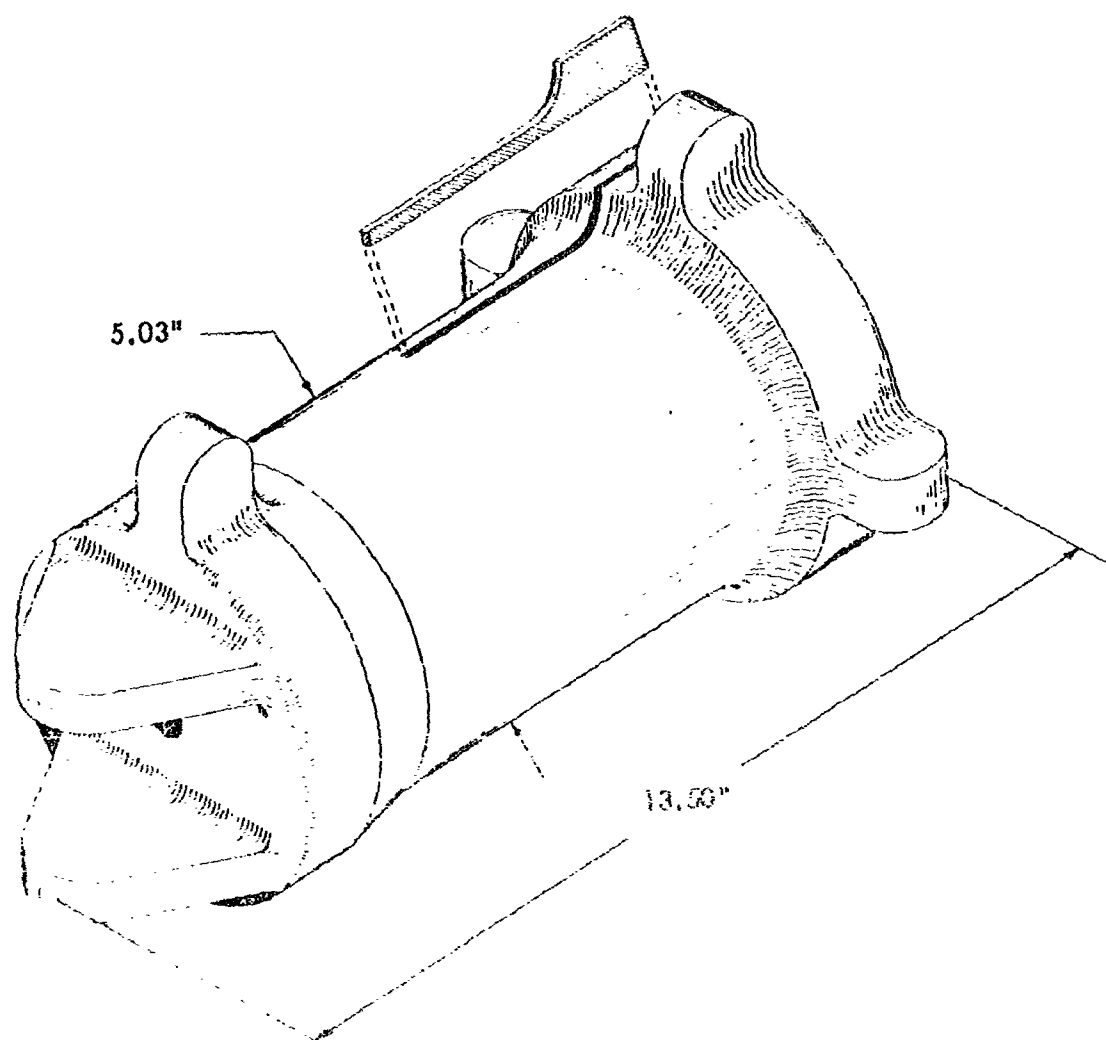
The Sway Brace Case was first cast from Ti-6Al-4V alloy in static molds of rammed graphite with rammed graphite cores. The casting was poured in the horizontal position, gating through four risers. A riser was located at each corner of the casting and connected to the casting through the lugs on one side and directly to the heavy section on the other side. Basic problems exist with this design for casting in titanium. It should be noted that minimum wall thickness throughout the cylindrical section of the part is 1/4-inch and that no difficulty was experienced in producing a smooth casting without misruns. However, as would be expected from the feeding studies, the uniformly thin center section had dispersed shrinkage porosity. As the section increased in thickness toward the ends of the casting and the riser locations, the shrinkage disappeared and the casting was sound with exception of the thermal centers adjacent to the lugs. Connection of the risers to the casting was difficult at these lug positions since an adequate connection would destroy definition of the lug outlines. Shrinkage occurred in the characteristic manner at the thermal centers behind the lugs, appearing as a single concentrated void existing at each location as shown in Figure D34B. It should be noted that the two tapered ears and the closed end of the casting were completely sound since their design produces exceptionally good directional solidification conditions.

Figure D34A shows the effect of pouring position on casting soundness. The casting was poured with the open end down. It was evident that gas was rising to the surface of the mold and being collected under the surface of the metal in all castings poured in this position. Shell cores were used for these trials. Risers were attached to the lugs extending from the parting line plane. The mold cavity was gated through each of the four riser locations. Shrinkage in the thin tube section was dispersed indicating poor feeding of the wall area. Heavy shrinkage was located at the lugs extending from the open end of the casting.

Figure D34B illustrates a casting that was poured with the open end at the top of the mold. The casting was gated through two of the four side risers with an additional four risers attached to the top of

FIGURE D32

SWAY BRACE CASE CASTING



the casting. Dispersed shrinkage was still present in the tubular section. The four top risers were attached to the top of the casting in an attempt to feed the lugs extending from this area. The riser connections were not sufficient to permit the metal to feed the top of the casting.

The initial pattern revisions were decided upon the basis of the experimental casting shown in Figure D34C. In this casting a heavy shell core was tapered from one end toward the open end of the Sway Brace Case to provide a thicker wall section near the top end of the casting. The core modification increased the feeding of the top section to the tube section. Shrinkage was still evident in the heavy top end, since no provisions were made in this modification for additional feeding of the heavy section at the top of the casting.

Figure D34D illustrates the modified casting with redesign and location of risers. The top riser is a wedge shaped ring covering the entire area at the top of the casting, to provide more efficient feeding of this area. The shell core was modified to include a 5° taper, starting at approximately 2 inches from the bottom of the core. This amount of taper was believed to be more than needed to establish proper directional solidification and soundness of the tubular section. The top section of the casting did not feed properly to obtain soundness because of the greater volume of metal located at the top of the open end of the casting.

The tubular section of the casting poured in this manner did not show any shrinkage. There was some surface roughness from both the rammed graphite mold and the machined graphite core. The core was machined hollow to 1/4-inch thickness at the open end to reduce the chilling effect and clean-out problems. There were a few cold shuts in the interior of the casting due to the rapid skin chilling effect of the machined graphite core. The shell core was not satisfactory for casting the Sway Brace Case because of the large amount of metal concentration. The shell core does not have sufficient heat capacity to prevent rapid heating to high temperature and consequent wetting of the core by the metal alloys, resulting in severe porosity. A casting setup for a centrifugal pump was made similar to that shown in Figure D35 in which a shell and tube casting was made. The shell and tube were made

relatively thick (1/2-inch) to give it sufficient thermal capacity to prevent penetration by the molten metal. Breakage of the core at the support hole occurred after the casting had solidified, without damage to the casting. Movement of the shell core, if any, was very slight.

The photograph in Figure D35 is of a casting setup in which a rammed graphite core was used. The use of a rammed graphite core with a machined graphite insert at the support end of the core gave consistently good results. The major disadvantage in the rammed graphite core is that it was difficult to remove from the casting. It was necessary to chip almost the entire core free from the casting. An advantage of the rammed core over shell core was the better dimensional accuracy in the finished castings.

The Sway Brace Casting was bottom gated through two side risers attached to the closed end of the part. These two risers provided feed metal for the two end lugs and the heavy portion of the casting to which they were attached. This gating arrangement provided uniform metal flow to the top riser, which was attached to the entire open end of the part, forming a ring riser. The inner portion of the Sway Brace Case was tapered from the open end to the closed end to provide a sound wall section. With incorporation of this taper, the wall section was consistently cast with very little shrinkage or gas defects. The ring riser at the top of the part supplied metal for the wall section and the heavy portion at the open end of the part. The part as centrifugally cast was consistently sound. The production run of 16 Sway Brace Cases was completed by pouring four heats of four parts each.

The parts were cast with 12 G's force on the mold cavity by centrifuging at 140 RPM. This speed was found to be necessary to obtain proper feeding from the ring riser attached to the top of the casting. The ring riser was relatively close to the spin axis of the mold setup and did not feed properly at reduced speeds.

The rejection rate for Sway Brace Cases was 25%. The cause of rejection was insufficient metal to fill the mold cavity. Casting yield was 28 per cent. There was no difficulty encountered in the pilot production casting of this part. The incorporation of taper in the bore

of the part provided adequate thermal gradients to properly feed the walls of the cylinder and heavy riserings in close proximity to thermal centers prevented shrinkage in the heavy sections.

FIGURE D34

Porosity in Sway Brace Case

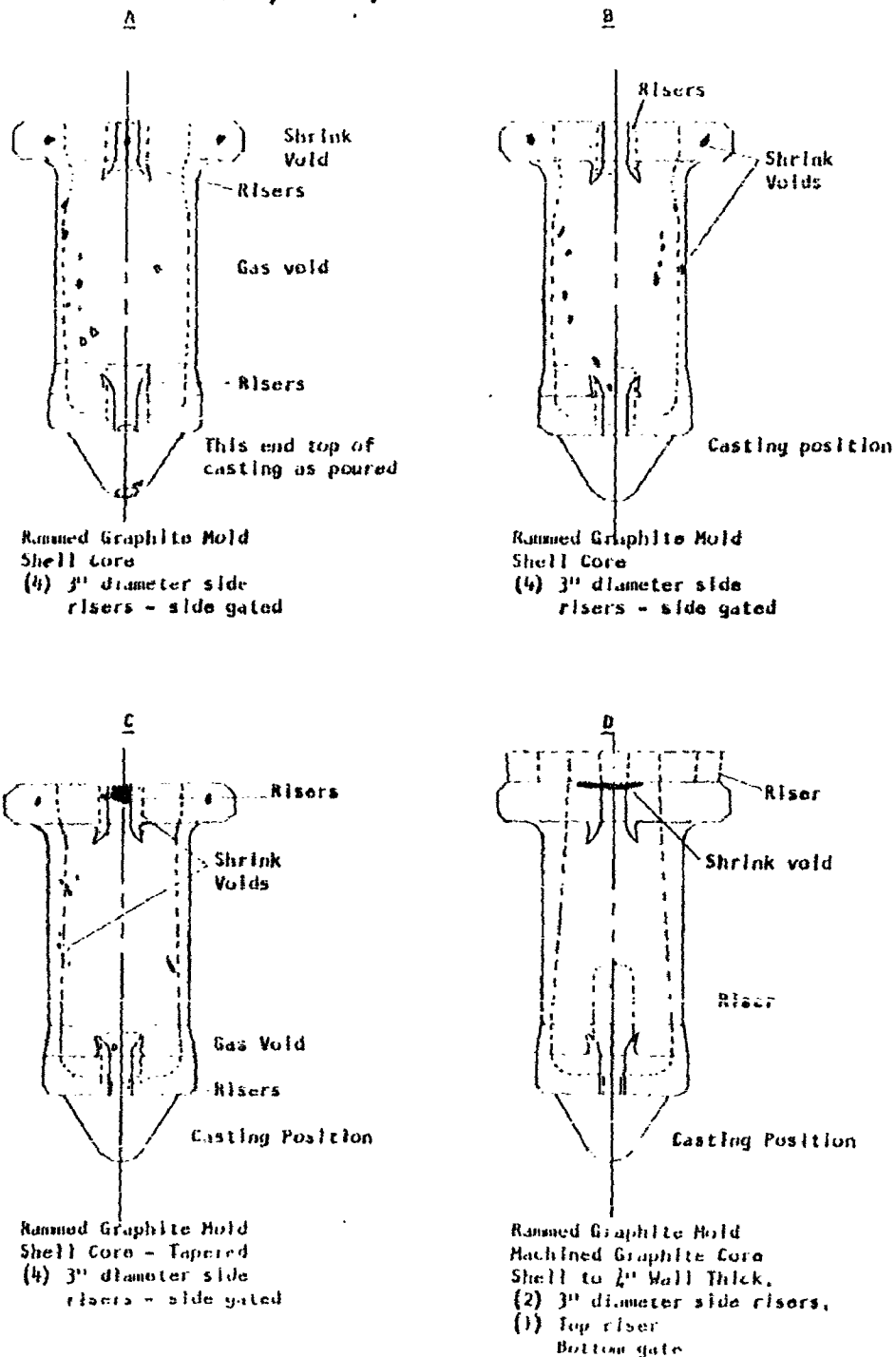
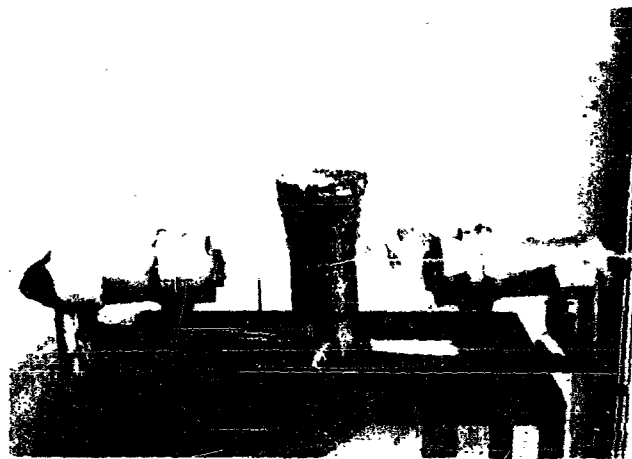


FIGURE D35

CENTRIFUGALLY CAST SWAY BRACE CASE

Rotated at 140 RPM (12 G)



Casting of the Elevator Torsion Fitting

The Elevator Torsion Fitting was first cast in machined graphite with a machined graphite core, in machined graphite with a rammed graphite core, and in rammed graphite. Machined cores produced the same cracking difficulty in the thin flat areas at each end of the casting as was experienced with the Bracket. This condition was readily corrected with the use of a rammed graphite core in the machined mold. As was expected from the result of the feeding studies, the casting evidenced shrinkage porosity in the wide thin webs at each end as shown in Figure D 38, and in the four heavy sections around the center of the part. Approximately two degrees of taper were added to the webs, thickening the casting toward the heavy center which decreased the total shrinkage but did not remove all shrinkage in that area.

The four heavy sections around the center of the Elevator Torsion Fitting presented difficulty in risering with sufficiently large riser connections, because the wall thickness of standard graphite tubing held the risers at least 1-1/4-inches apart, and the slight jog at the edge of the casting prevented a full-width riser connection without loss of casting detail. Characteristic shrinkage in these heavy sections is shown in Figure D 39.

Through further casting attempts, it was determined that the web section with taper to approximately 1/4-Inch of the center did not have sufficient taper to promote directional solidification. In attempting to obtain sound web sections, risers were located along the edges of the channel to feed the web area. This was not successful. The thin sections required feeding to remove dispersed shrinkage, but were not thick enough that risering at the edges of the channels would provide the directional solidification required to remove the shrinkage.

The next trials in statically casting the Torsion Fitting were based on increasing the taper of the thin web section and increasing and modifying gating and risering.

FIGURE D36

ELEVATOR TORSION FITTING

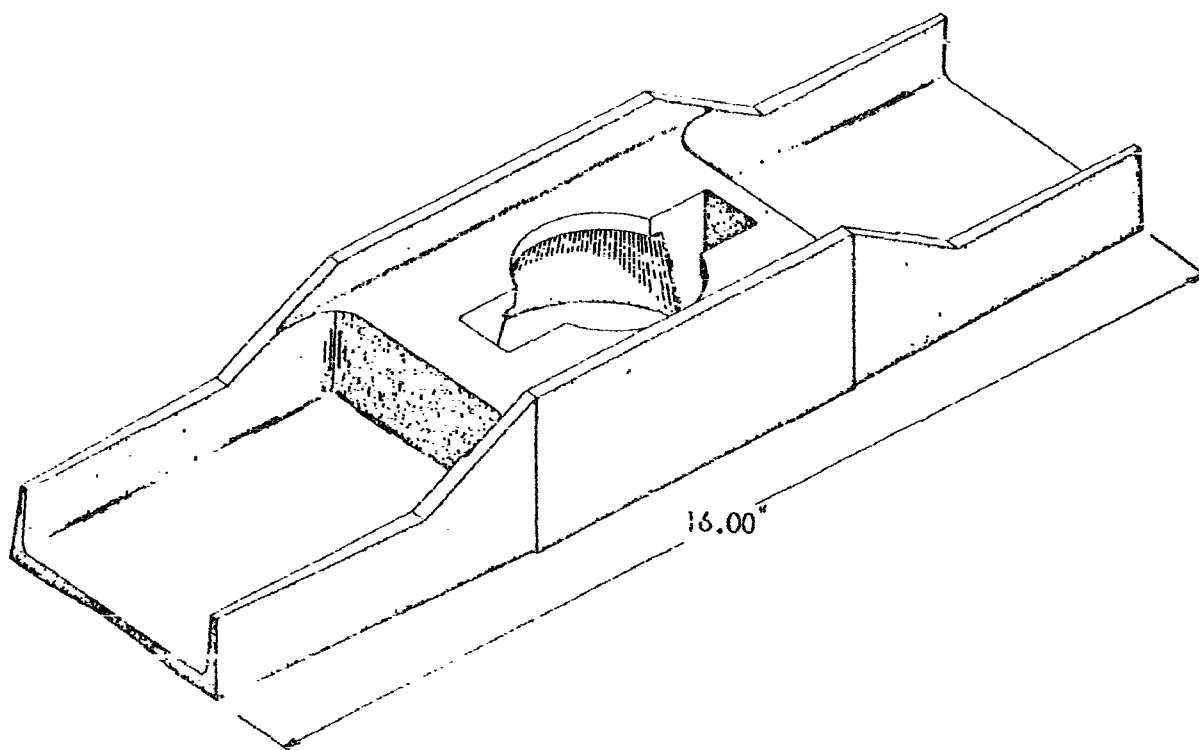


FIGURE D37

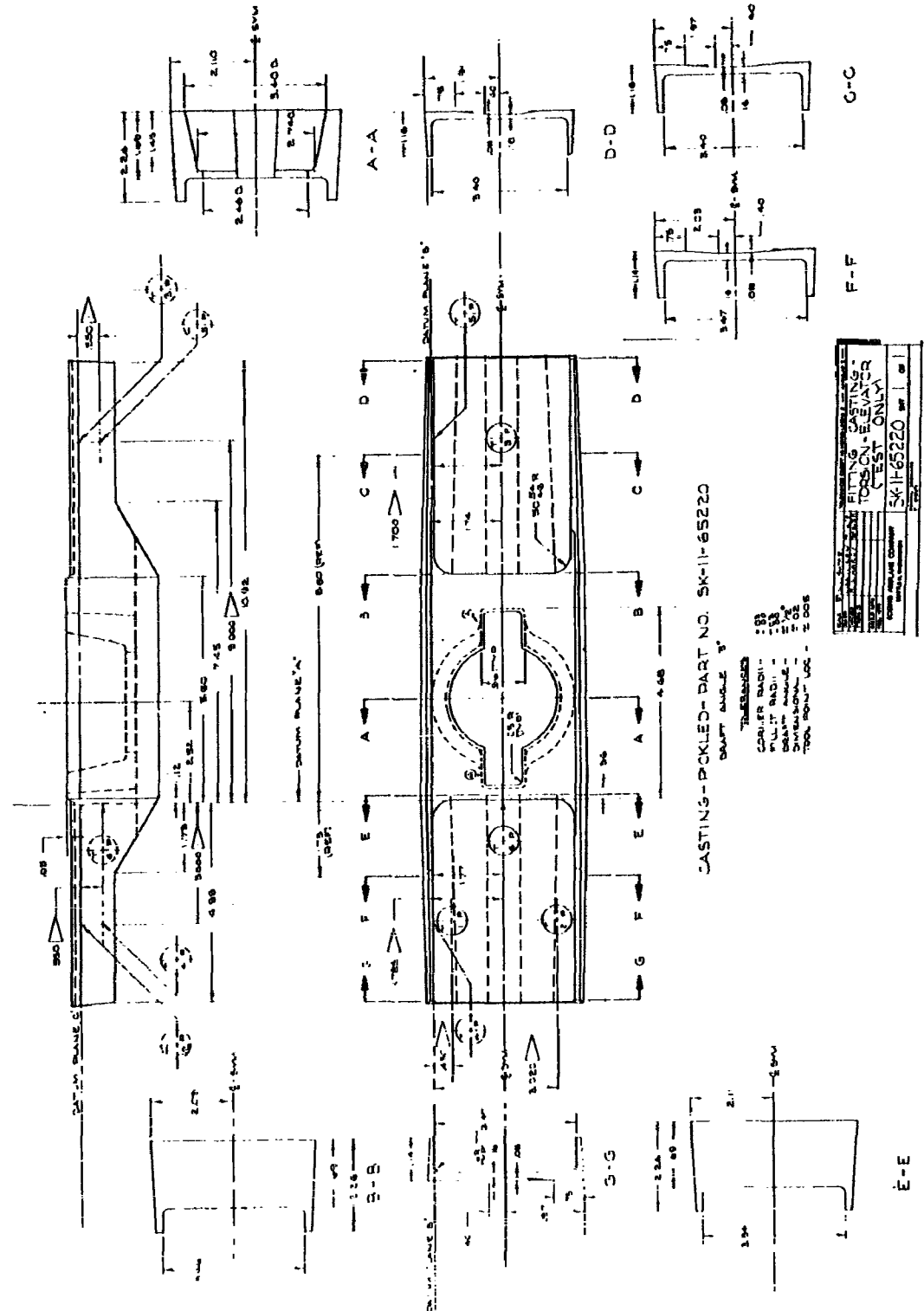


Figure D-40 C and D-40 D castings were poured to determine if an advantage could be gained by pouring the casting upside down from its normal position. Figure D-40 C, poured in the usual position, illustrates the considerable improvement in the web area with one end being poured into the mold. Figure D-40 D, poured upside down, shows the same section and dispersed chills placed in the web area. The difference in the two castings is that the casting poured in the usual position (Figure D-40 C) has a large area of porosity in the web area, while the casting poured upside down (Figure D-40 D) has a much smaller area of porosity in the web area.

The photograph in Figure D41 illustrates a static casting setup with side risers along the tapered web area. The web area had been revised to include a 14° taper from the center of the web to the edges in an attempt to obtain a sound casting. The addition of 14° of taper to the web section did not provide adequate thermal gradients to obtain soundness. Adding taper from the center of the web section to the edge did not successfully remove shrinkage in any of the trials conducted. The heavy rectangular riser shown located at the central portion of the casting, feeding the heavy section, did not result in a sound heavy section.

The photograph in Figure D42 is of an experimental pour made to determine the effectiveness of adding taper from the extreme ends of the web towards the center section. The taper added to the web section was parabolic, starting at 7° at the extreme ends of the casting. The tapers required to feed the section of the casting were determined from the feeding distance studies; i.e. for the first inch from the end a 7° taper is required to cast a one-inch length of $1/8$ -inch section thickness sound or to obtain a soundness ratio of eight. The second taper was 6° to the two-inch length and thereafter 2° to the five-inch length.

The X-ray of the casting did not reveal any dispersed shrinkage, although some gas porosity was apparent. This type of taper was more effective in increasing the feeding distance than a straight taper. Upon straight tapering 6° from the end of the casting to the heavy center section, dispersed shrinkage occurred at the outer edges of the thin web section. The parabolic taper not only increased sound feeding distance but reduced the total volume of metal required to pour the casting. As experimental casting trials progressed, experience showed a necessity for large tapers on large thin sections and less taper as section thickness increases. Several factors involved in the cooling of the metal contributed to this. The rapid chilling effect of the graphite mold causes thin sections to freeze very rapidly, and as section thickness increases heat transfer is less rapid. The problem is similar to all casting processes but is exaggerated in the case of titanium because of high melting temperatures and high thermal conductance of the mold materials.

The casting was made centrifugally in rammed graphite molds. Figure D 43 A is a sketch of the first centrifuge cast part. Some dispersed shrinkage was prominent in one of the web areas while the opposite end had slight amounts of shrinkage close to the heavy center section. Isothermal areas under the ingates presented two shrinkage areas at the same end of the heavy section that contained the greatest degree of shrinkage in the web area. The part was gated at the four heavy sections at the center of the casting. The four gates were attached perpendicularly to the casting and centered sprue. The castings were in a horizontal position attached centrally about a sprue. Castings poured in this arrangement did not give satisfactory results in surface quality because of excessive turbulence as the metal enters the mold.

The sketch in Figure D 43 B represents a casting centrifuged using the pour technique shown in Figure D 44. The molds were side gated and cast in the horizontal axis centrifugal casting furnace. The benefits of gating in this manner were reduced turbulence and continued pressure on the mold cavity as solidification took place. The thin web section was relatively free of general dispersed shrinkage, although gas porosity was evident in this section.

Figure D 43 C is an example of a casting containing large amounts of gas porosity. This casting was made by tapering the web section to obtain a 14° taper from the center of the web toward the outside edge and the rectangular riser located to feed the web area. The gas porosity may have been intensified by machining the rammed mold and exposing the subsurface of the rammed graphite mold. The center section was free from shrinkage. The casting was side gated through the risers located at the heavy center section of the casting.

Figure D 43 D illustrates the casting modified to incorporate 6° of taper from the end of the casting at the web area toward the center section. Risers were located on the top of the casting and overlapped the sides with bottom gating employed to feed the casting. Shrinkage was greatly reduced, but was still present both in the web area and at the heavy center section.

Absence of gas porosity in centrifugally cast parts was noted when molds were centrifuged at high speeds. Localized shrinkage at the heavy thermal areas in the center of the casting illustrated that sufficient metal supply was not available to feed those areas as solidification progressed.

The sketch in Figure D 45 shows the characteristic dispersed shrinkage found in the part when cast with inadequate padding and centrifuged at high speeds. The casting shown was centrifuged at 140 rpm (12 G's). It has been established through experimental heats that a higher G force will not remove this shrinkage. The ultimate removal of shrinkage such as this depends almost entirely on improved padding and risering.

The sketch in Figure D 46 shows the location of shrinkage when a 1 - 1/2-inch diameter riser has been added at the center of the web. The addition of this riser improved the soundness of the part by moving the shrinkage area to within 1/2-inch of the ends of the part. The illustration in Figure D 47 shows the location of shrinkage with a 2-inch diameter riser used. The shrinkage still remains near the ends of the web.

Risering of the center of the web section does not appear to be an effective method of obtaining a sound section. A thin web does not lend itself to feeding from a centrally located riser, as has also been shown in previous feeding distance studies on tapered plates. The present casting attempts also demonstrate that the feeding distance of a centrally located riser is not influenced greatly by centrifugal casting techniques.

Figure D 48 illustrates a further modification of the risering system. A riser of large connecting area is at the end of the web at the location of previous dispersed shrinkage. Occasional shrinkage occurred such as shown in the sketch.

Two heats were required to make the pilot production run of ten castings. Six molds were poured in each heat.

The gating and risering system was similar to that shown in Figure D 48. The molds were centrifuged to obtain 12 G's on the mold cavity, requiring centrifuge speed of 140 rpm to obtain proper casting conditions.

The rejection rate for the castings was 17%. The two casting rejections were caused by breakage of mold setup braces and consequent dislocation of molds during pouring. The yield of casting weight to poured weight was 20%.

The Torsion Fitting was the most difficult part to cast to the internal soundness levels required. The thin web section is very difficult to feed properly to establish thermal gradients required for casting sound parts. In some instances a random dispersed shrinkage was located at the extreme ends of the fittings. The 12 G force on the mold cavity has succeeded in eliminating the shrinkage occurring in most instances.

FIGURE D38

DISPERSED SHRINKAGE IN WEB SECTION --
MACHINED GRAPHITE MOLD

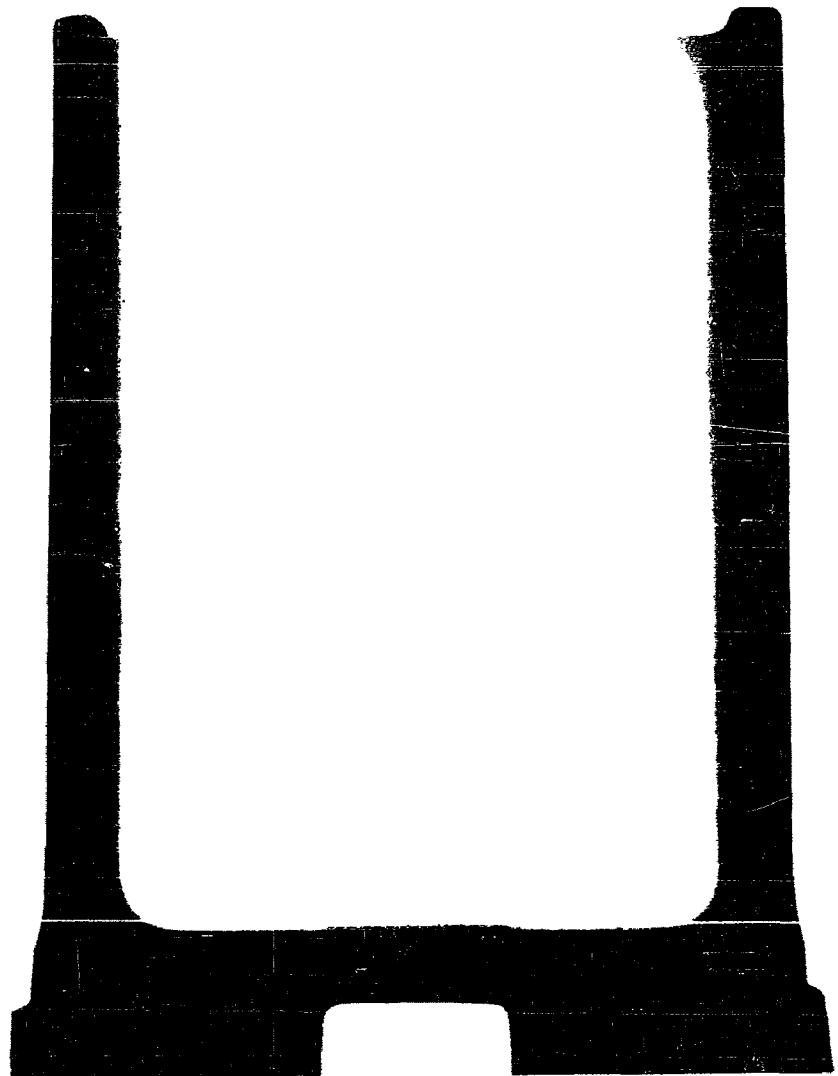


FIGURE D39

SHRINKAGE IN HEAVY SECTION

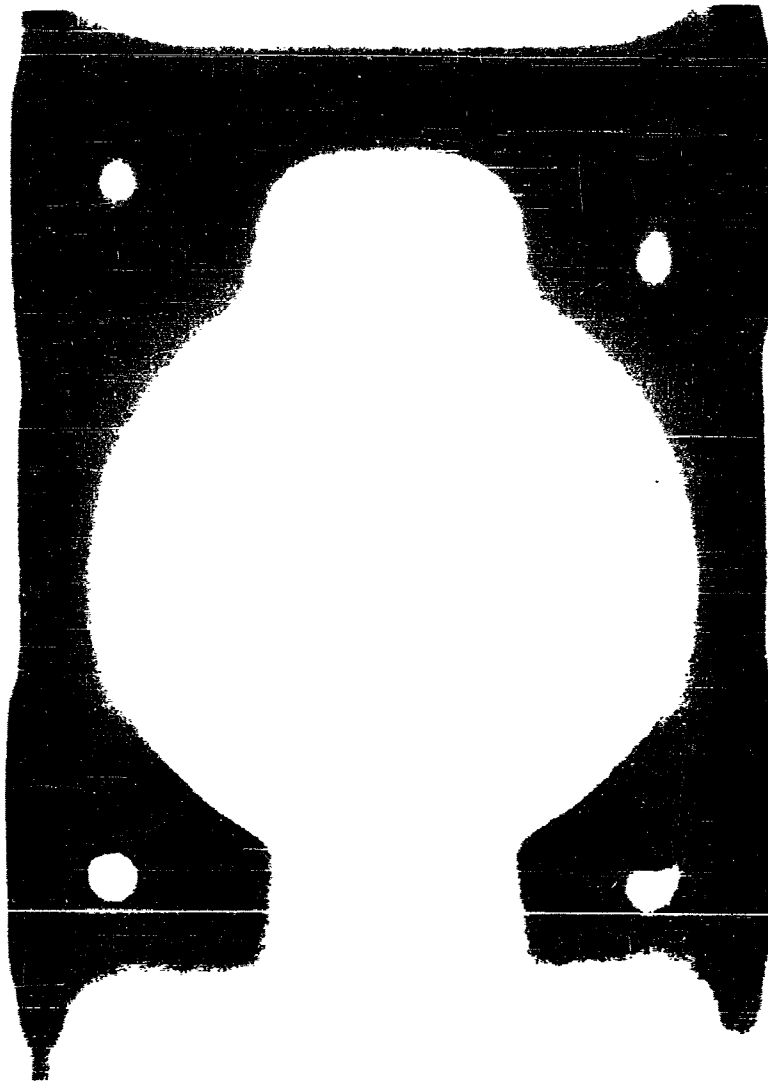


FIGURE D40

Porosity in Torsion Fitting

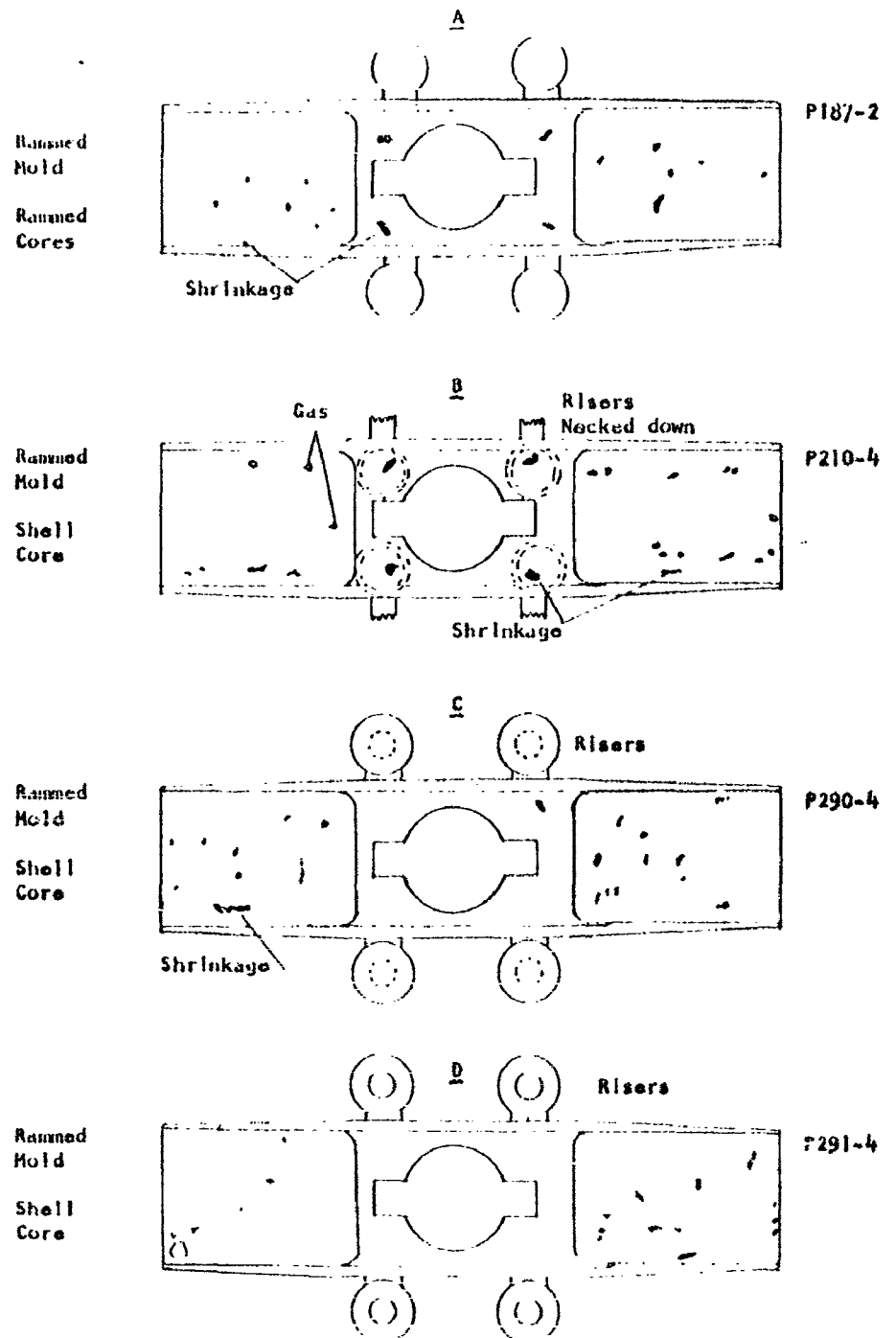


FIGURE D47

**STATICALLY CAST TORSION FITTING**

Rammed Graphite Mold, Shell Core

Result below

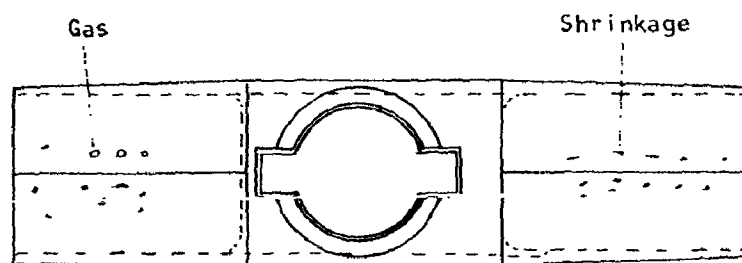
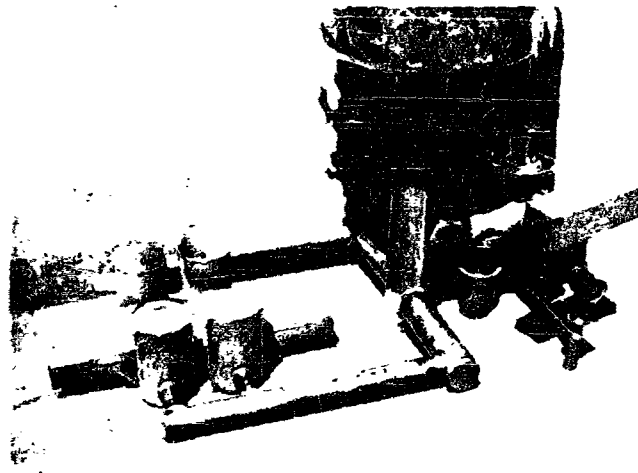


FIGURE D42



STATICALLY CAST TORSION FITTING

Rammed Graphite Mold, Shell Core

Result below

Gas Porosity

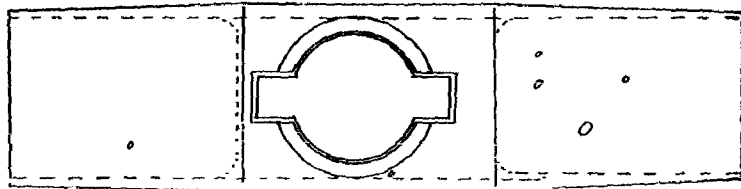


FIGURE D43

Porosity In Torsion Fitting

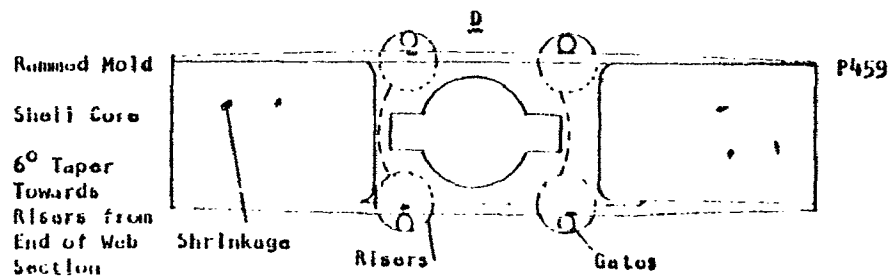
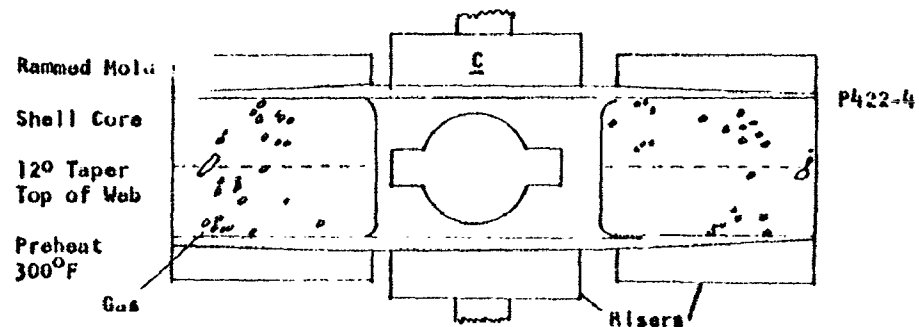
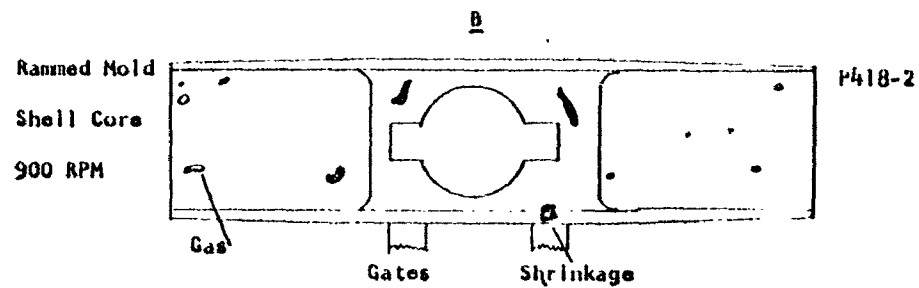
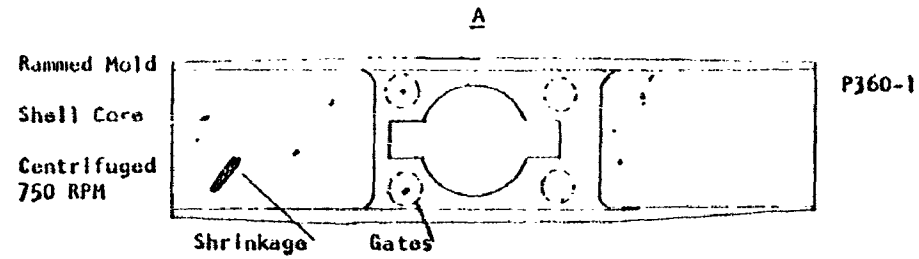


FIGURE D44

TORSION SUPPORT FITTING CENTRIFUGALLY CAST
IN RAMMED GRAPHITE MOLD

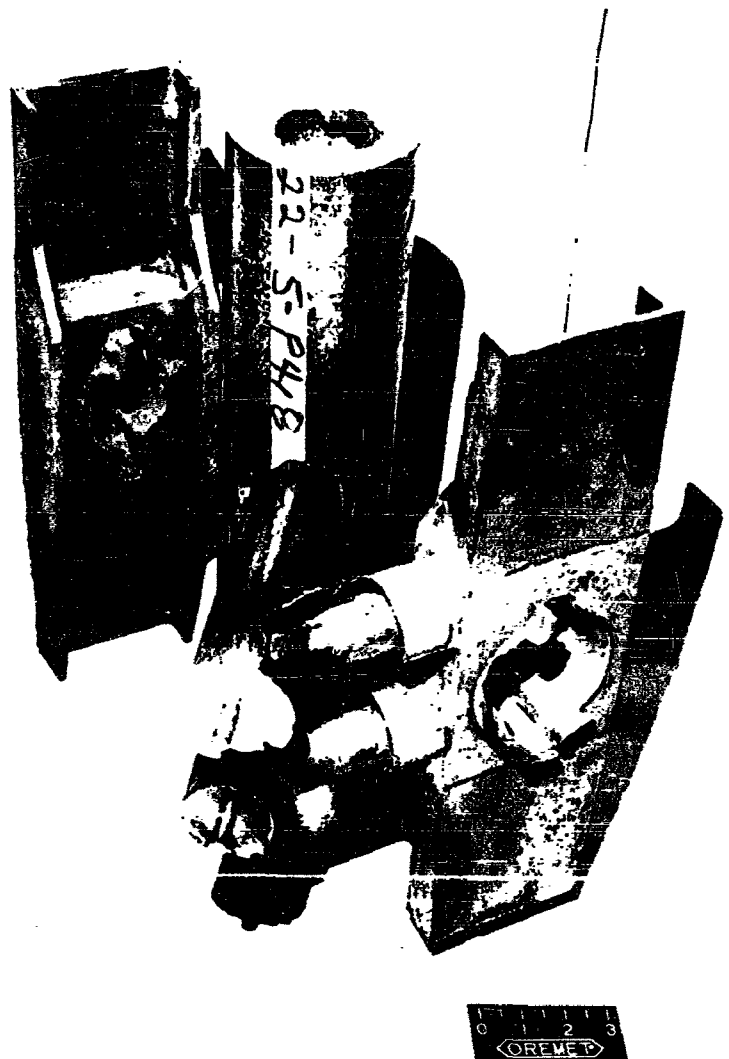
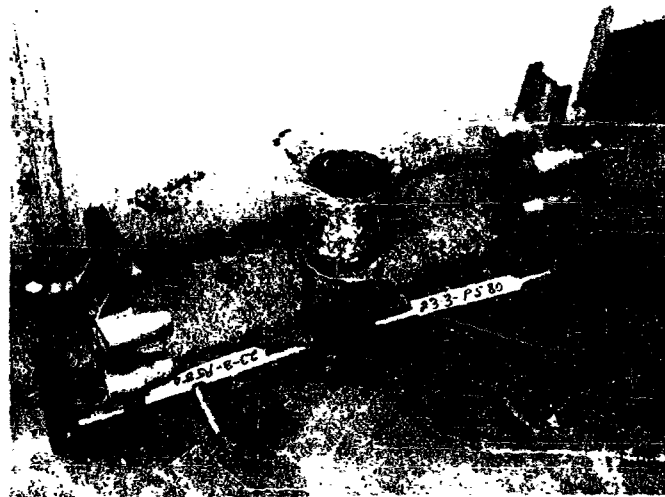


FIGURE D45

CENTRIFUGALLY CAST TORSION FITTING

Rotated at 140 RPM (12 G)



Shrinkage

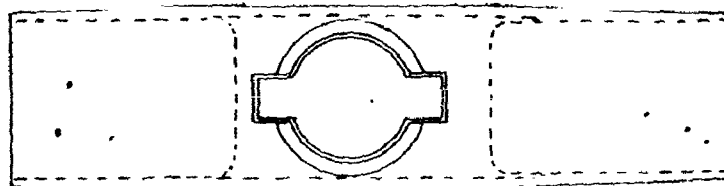
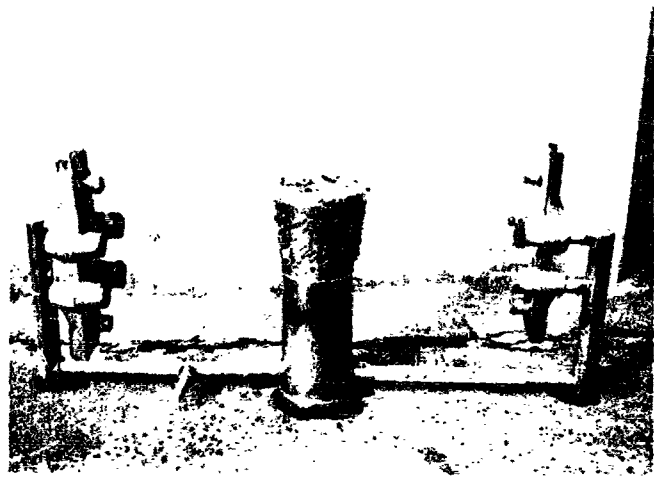


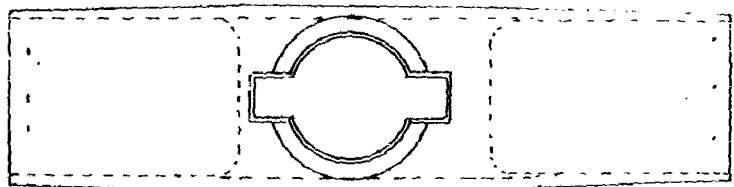
FIGURE D46

CENTRIFUGALLY CAST TORSION FITTING

Rotated at 160 RPM (15 G)



1 1/2-inch Diameter Riser Added to Centers of Webs

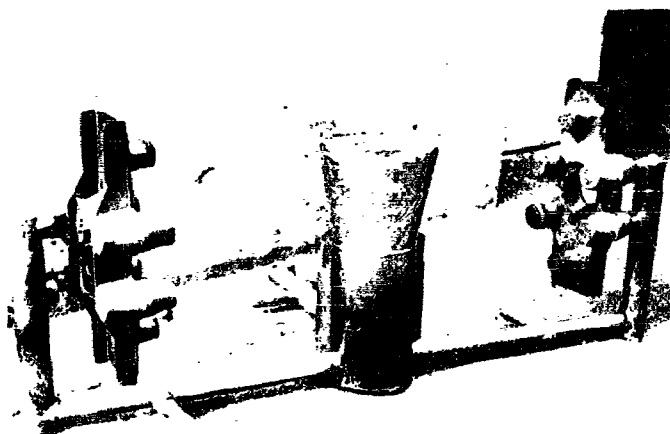


Shrinkage

FIGURE D47

CENTRIFUGALLY CAST TORSION FITTING

Rotated at 160 RPM (15 G)



2-inch Diameter Riser Added to Centers of Webs

Shrinkage

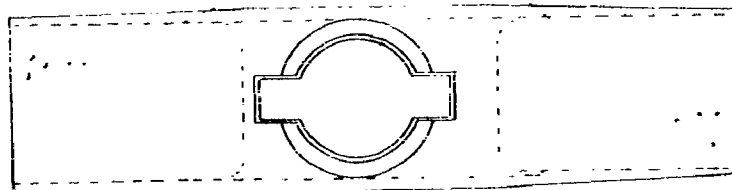
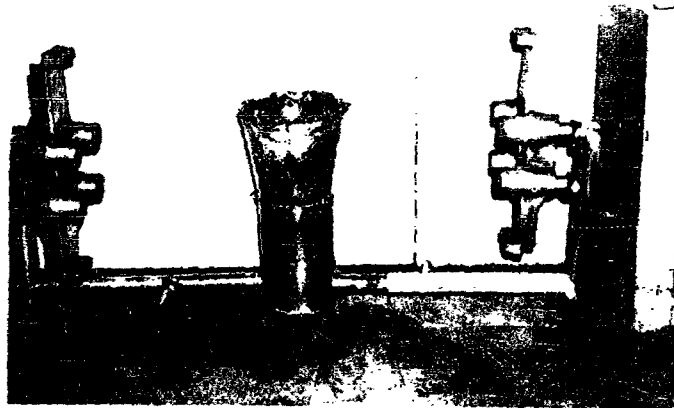


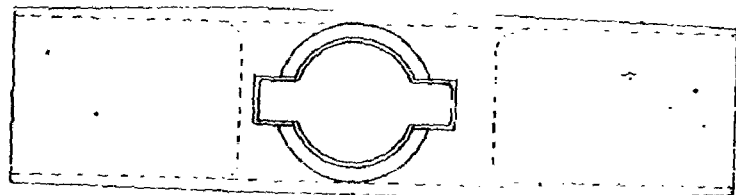
FIGURE D48

CENTRIFUGALLY CAST TORSION FITTING



Rectangular Riser Added to Ends of Webs

Shrinkage



Casting of the Horizontal Stabilizer Fitting

The Horizontal Stabilizer Fitting was first cast statically. The mold required sixteen sections fitted together. Gating and risering was accomplished by drilling into the rammed graphite mold and by addition of graphite piping to provide the external connections.

This casting was not sound. Numerous gross shrinkage areas occurred as a result of insufficient risering. The web areas of the part contained widely dispersed shrinkage demonstrating that inadequate tapers had been incorporated in the original pattern equipment. In addition to the shrinkage problem, severe gas porosity was noted.

The photograph in Figure D50 shows the front of the casting with gates and risers attached. In this casting experiment two-thirds of the metal was theoretically to flow through the large front gate and through its interconnecting system, feeding the length of the fitting. The remaining one-third was to flow through the two smaller gates into the enclosed arm of the part, thereby preventing cold shuts and surface laps. The system succeeded in minimizing surface cold shuts, but not in entirely eliminating them. A slight misrun occurred in one of the thin webs in the bottom of the mold as shown in Figure D51. This misrun was due primarily to too much metal flow through the end ingates and not through the next inside ingate. It would be necessary to reduce the size of the runner or ingate feeding the end of the casting to provide a more desirable flow of metal.

Figure D51 illustrates the bottom of the casting and the cored out detail. Evidence of blowing is visible in the web areas of the cored sections. The cause of this irregularity was not determined.

The second attempt to cast the Horizontal Stabilizer Fitting was accomplished in the vertical axis centrifuge furnace. The part was extremely large for centrifuge casting in this furnace.

This casting also was not sound. It contained numerous shrinkage defects where risering was insufficient. There was a noticeable increase in soundness of the web areas of the angular 'U' section and the extremities of the Fitting. The center portion of the casting was absent, due to insufficient metal in the pour. The casting was gated along the trailing edges, to provide low turbulence metal flow into the mold

cavity. The mold cavity was gated at several places from a main runner, and gating size changed to provide uniform filling through each of the gates. The 'U' shaped extension was also gated in three locations at the trailing edge to fill this area at approximately the same time as filling of the remainder of the mold cavity. It was expected that filling of the center portion would be difficult, and therefore a large gate was provided to feed metal to the center. Because of spilling and splashing there was insufficient metal to fill the mold cavity.

It was concluded that although sufficient development would be expected to solve the shrinkage porosity problem, the gas porosity problem could not be overcome without centrifuge casting the part. This component was too large for centrifuge casting in the furnace equipment available.

FIGURE D4.

INBOARD BEARING SUPPORT RIB, HORIZONTAL STABILIZER
(as machined from forging)

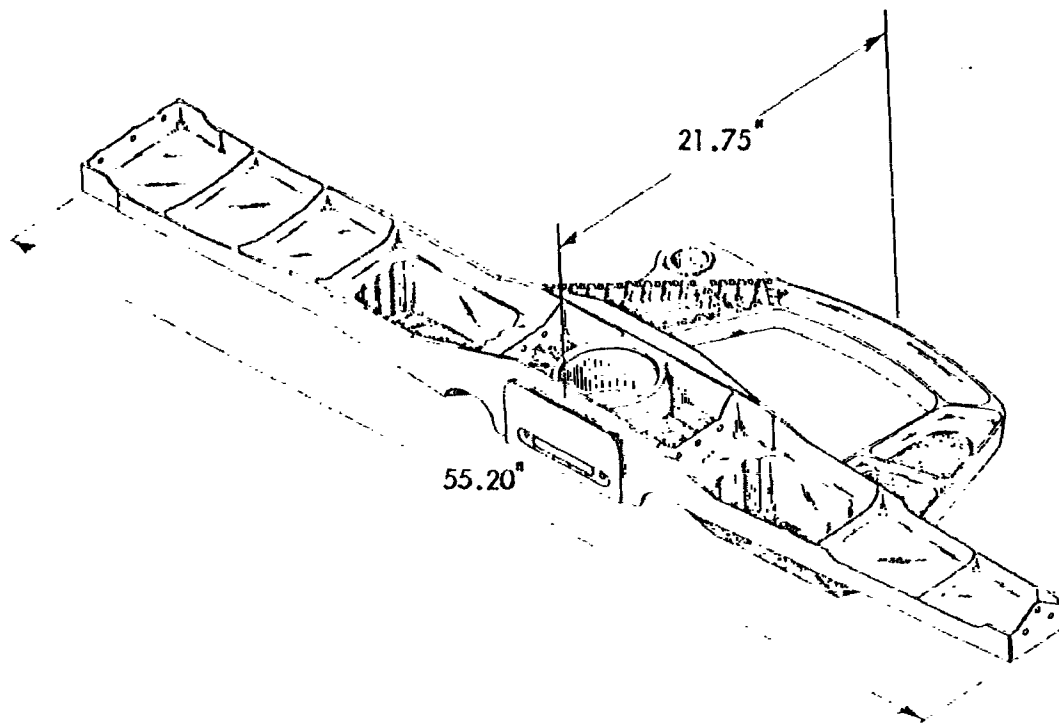


FIGURE D49A

TYPICAL SECTIONS - INBOARD BEARING SUPPORT RIB

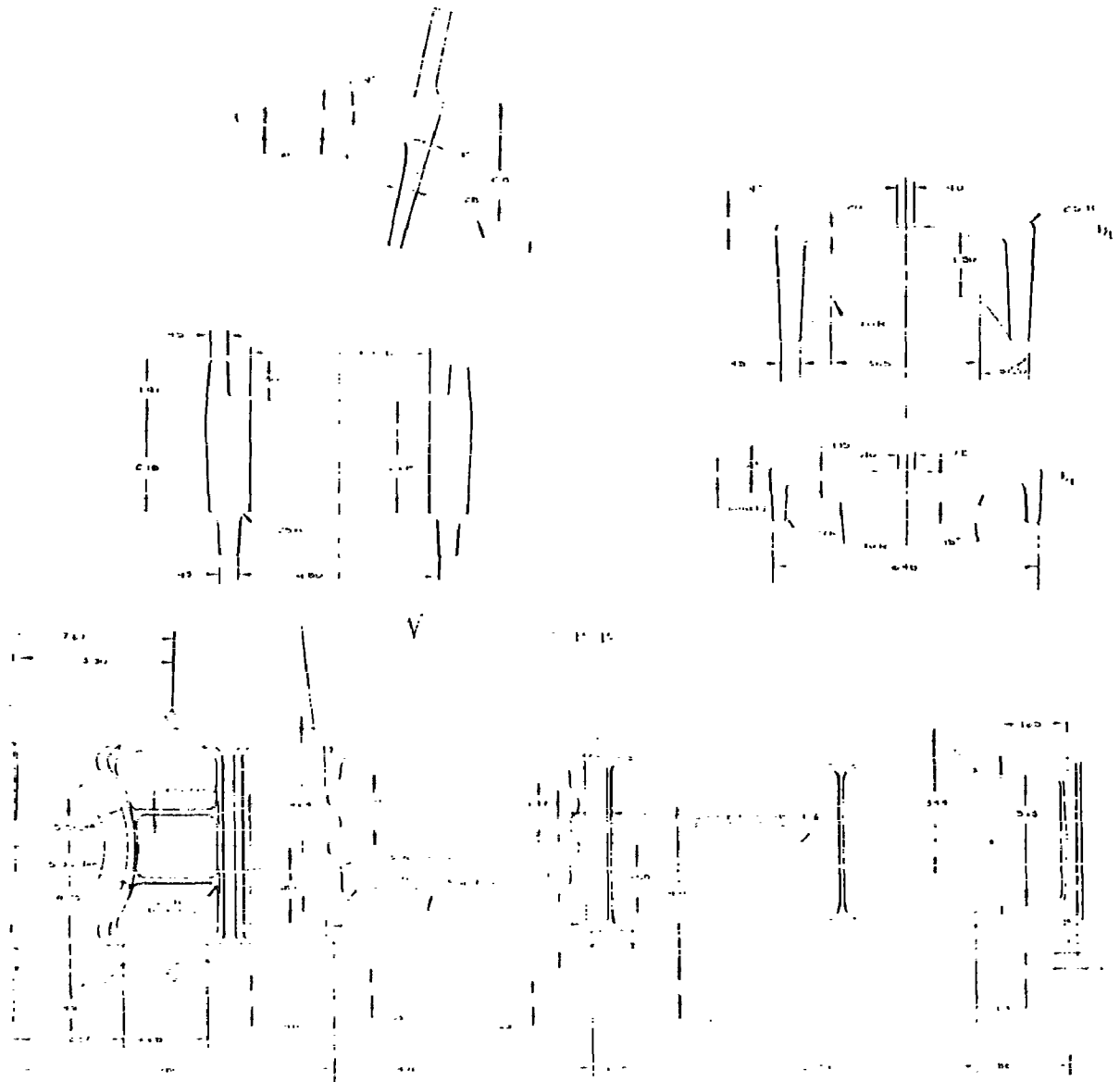


FIGURE D50

STATICALLY CAST HORIZONTAL STABILIZER FITTING

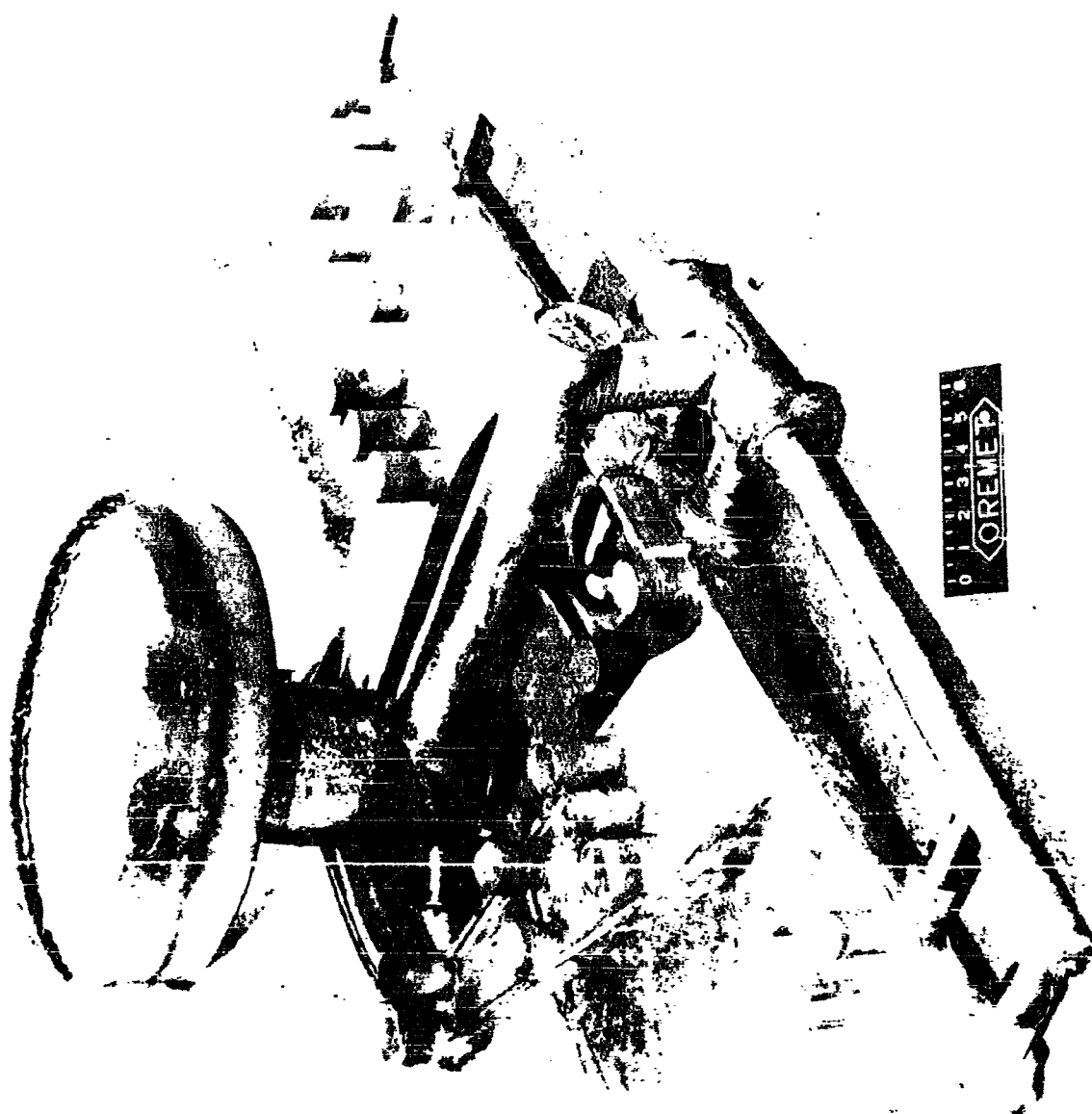
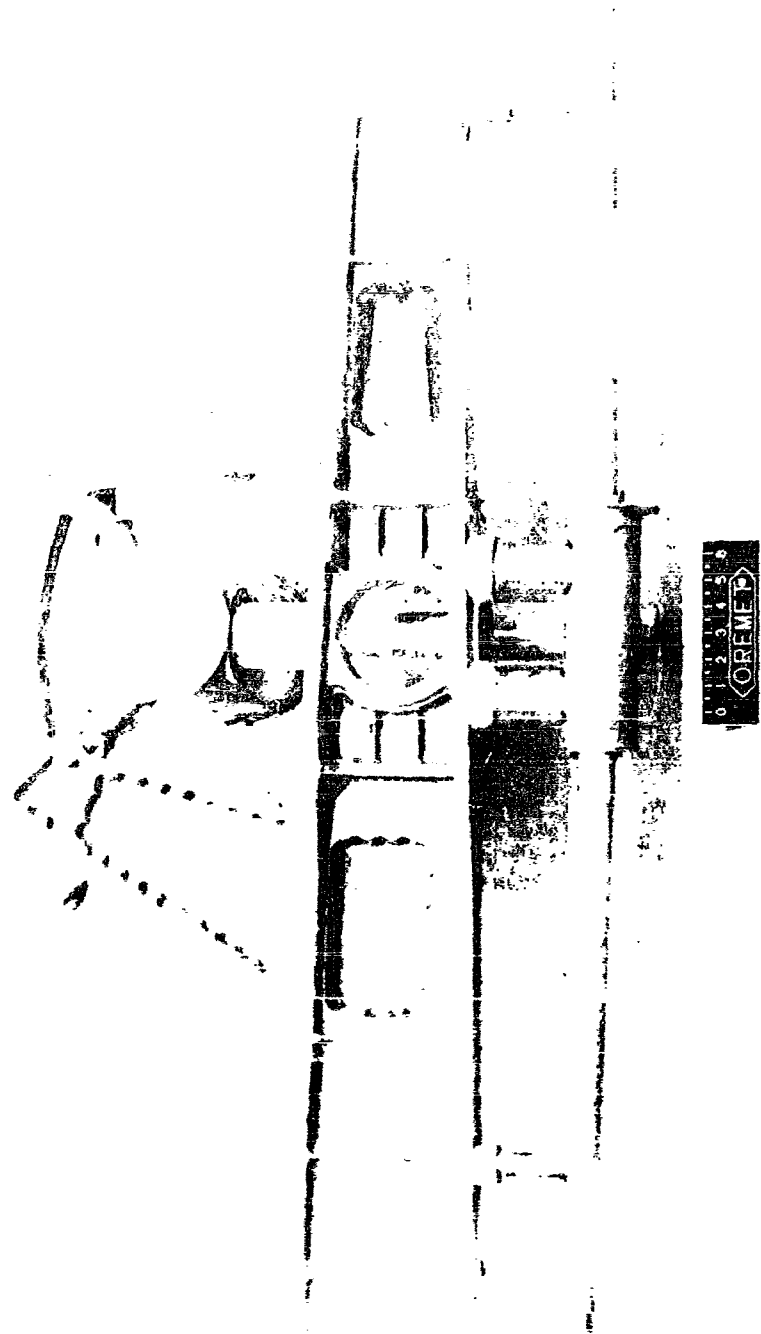


FIGURE D51

STATICALLY CAST HORIZONTAL STABILIZER FITTING



D2-2786-8

E1

SECTION E

EVALUATION TESTING

The following mechanical tests were used to evaluate the general properties of cast titanium alloys and to compare the structural components with conventional materials and fabrication methods:

(1) Conventional Materials Tests

- (a) Tensile, room and elevated temperature
- (b) Compression
- (c) Impact
- (d) Shear
- (e) Tension fatigue
- (f) Notched tensile
- (g) Bearing

(2) Component Tests

- (a) Developmental Bracket - Static and fatigue simulated service tests, cast Ti-6Al-4V compared to cast type 410 steel.
- (b) Flap Track Link - Static and fatigue simulated service tests, cast Ti-6Al-4V compared to forged AISI 4340 steel.
- (c) Elevator Torsion Fitting - Torsion load vs deflection simulated service tests, cast Ti-6Al-4V compared to AISI 4340 hog-out.
- (d) Sway Brace Case - Static simulated service tests, cast Ti-6Al-4V compared to cast type 410 steel.
- (e) Inboard Bearing Support Rib - No testing do to failure to develop a satisfactory titanium casting.

Conventional Materials Tests

All tensile tests were conducted on conventional 1/4-inch gage diameter, 1-inch gage length tensile specimens similar to those specified in Federal Test Method Standard No. 151, type R3 specimen. The strain

rate used was $.005 \pm .002$ in./in./minute through 0.2 per cent strain. All tensile yield strengths reported are at 0.2 per cent offset. The results of tension tests are reported in the section of this report on alloy development.

The notched tension test specimens were as shown in Figure E1 and had a notch factor (K_t) of 3.3. Results of notched tests were used to compare properties of experimental alloys and are reported in the alloy development section.

The impact test specimens were conventional charpy test specimens. Results of impact tests are reported in the alloy development section.

The compression, shear, and bearing tests were on conventional specimens and were conducted to provide allowables data for design purposes. Results are in the section on development of specifications, standards, and design allowables.

Tension fatigue specimens were tested to establish requirements for chemical removal of surface contamination. The results of the tests are reported on page B 60.

COMPONENT TESTS

Testing of The Developmental Bracket

The tests on components were to compare cast titanium structural shapes with the presently conventional aircraft materials and fabrication methods. All tests were at room temperature and were designed to simulate service conditions applicable to each part. All parts tested were compared with the present production part, except the developmental bracket which was not a production part. Two attempts were made to produce this part as a titanium forging, but the thin webs and high thin flanges made the part impractical to forge. Therefore, type 410 steel castings were obtained and tested for comparison with the titanium castings.

The Bracket was prepared for testing by milling both faces of the two heavy lugs and a slot in the small lug and then drilling and reaming the fastener holes. Castings were considered satisfactory for testing if X-ray defects were absent in all areas except the web and the two flanges which received almost no load. In those areas, slight shrinkage porosity was accepted. The castings were fatigue tested using the jig shown in Figure E2 and static tested in a conventional tensile test machine using a similar jig. The castings were loaded 26.6° from vertical (as shown in Figure E2) to provide a horizontal load equal to one half the vertical load.

All of the type 410 steel castings were X-ray sound and were heat treated by the vendor to 180,000 psi minimum ultimate tensile strength, 150,000 psi minimum yield strength, and 25 per cent minimum reduction of area. The Ti-6Al-4V castings were in the as-cast condition and had not been pickled.

Four cast Ti-6Al-4V brackets and one type 410 steel bracket were statically loaded to failure. The four cast Ti-6Al-4V brackets failed at the mounting hole in the upper large lug (Figure E3) at 24,400, 27,900, 25,000, and 21,800 pounds load. The type 410 steel bracket failed in the upper flange (Figure E3) at 34,100 pounds load. The average of the four cast Ti-6Al-4V values is 73 per cent of the ultimate load for the type 410 steel bracket. This value is very close to the ratios of ultimate tensile strengths of the two materials.

Ten cast Ti-6Al-4V Brackets and seven cast 410 steel brackets were fatigue tested for comparison of fatigue properties. Table J23 and Figure E4 show the results of these tests. The fatigue strength of the Ti-6Al-4V castings at any given number of cycles is 55 to 60 per cent of that of the 410 steel castings.

An additional series of fatigue tests was conducted on a modified version of the Bracket to determine the necessity of chemical removal of surface material from the as-cast surface. Since several of the brackets already tested failed adjacent to the machined mounting hole, in the upper lug, that lug was thickened by twenty per cent to make the casting fail in the flange, which had an as-cast surface. In addition the flange was tapered from its midpoint toward the lugs to encourage failure at the flange midpoint. Although the static strength of the modified design was an average of 37,000 pounds load for three tests (compared to 25,000 pounds for the unmodified design), the casting continued to fail frequently at the same machined hole. For this reason, a tension fatigue specimen having a gage diameter of 0.5 inch was designed and used for the chemical removal study. The results of this study are reported on pages B59 and B60, in Figure B30, and Table J16.

FIGURE E1

NOTCHED TENSILE TEST SPECIMEN
($K_t = 3.3$)

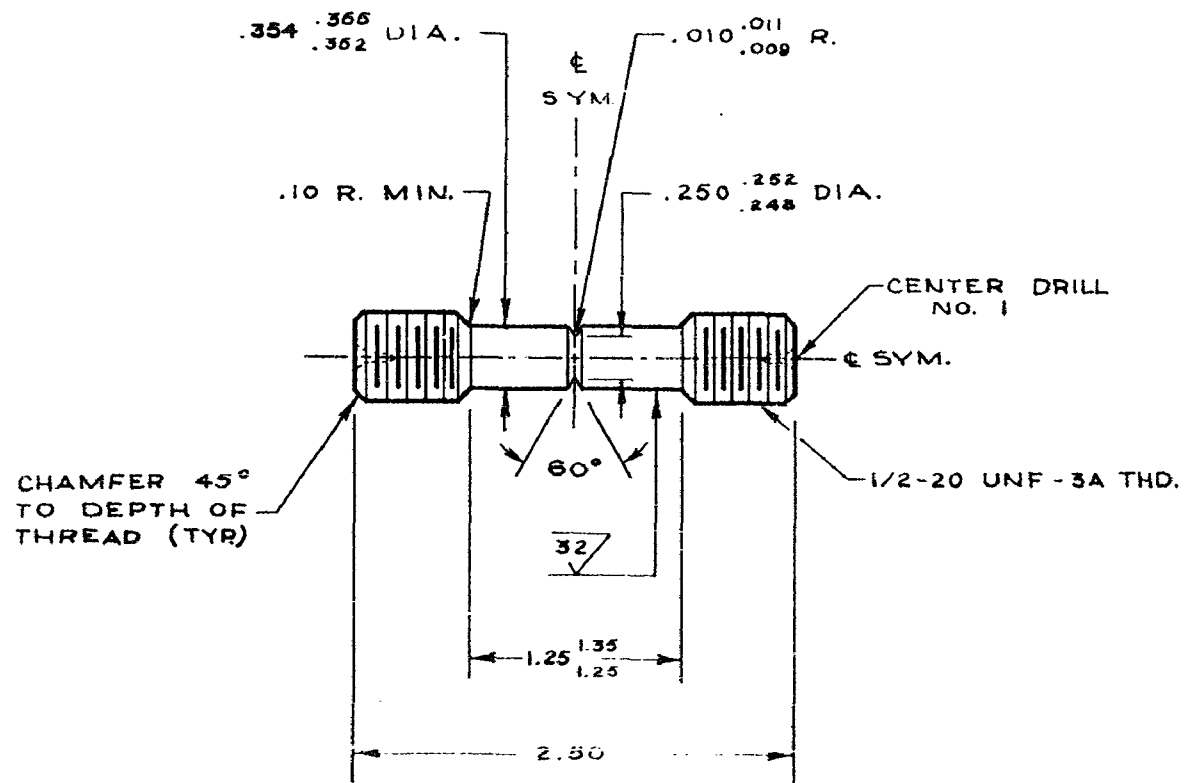


FIGURE E2

SETUP FOR FATIGUE TEST OF DEVELOPMENTAL BRACKET

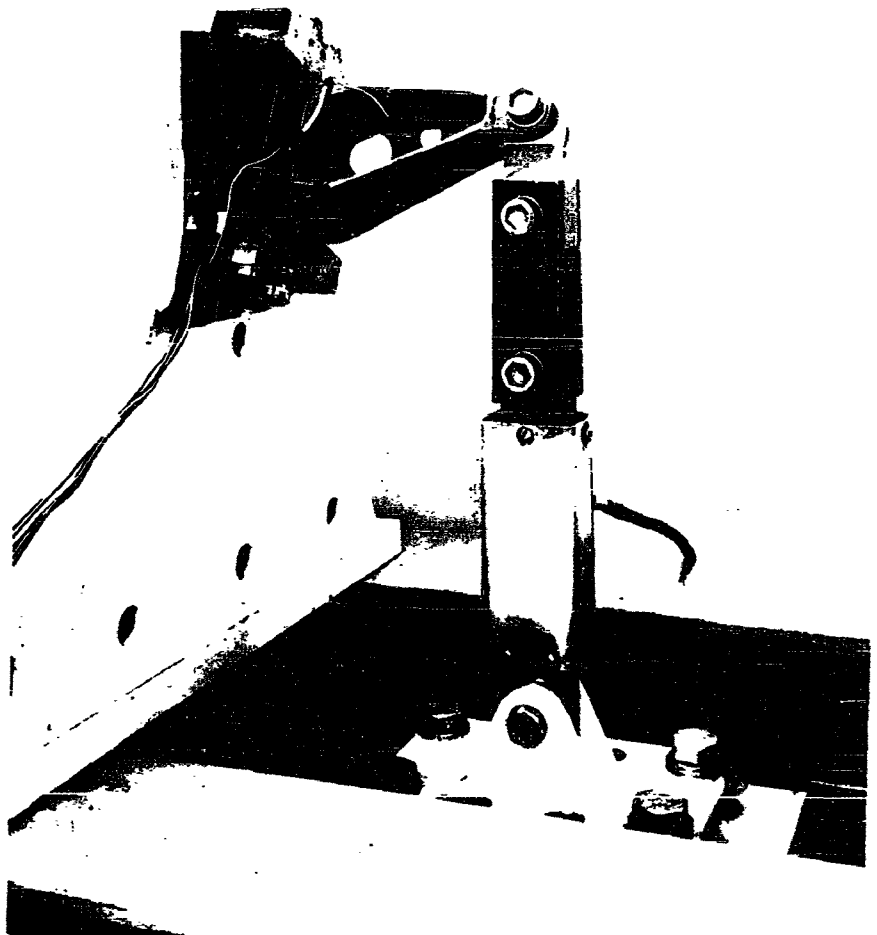


FIGURE E3

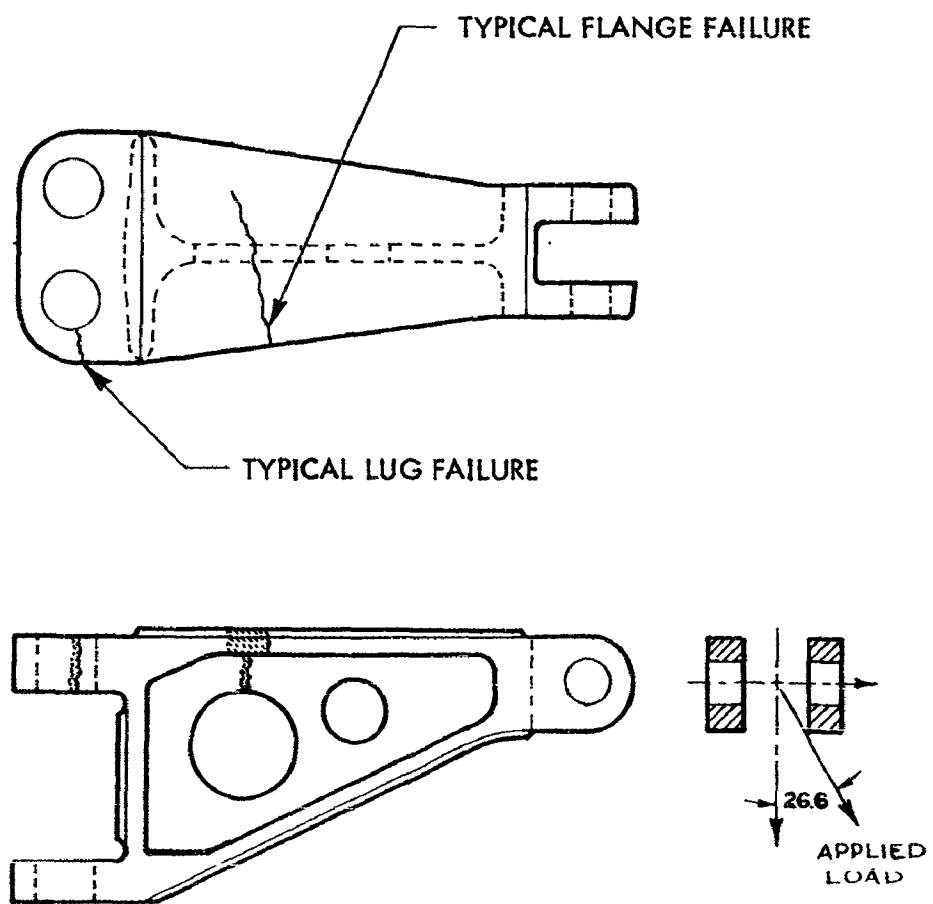
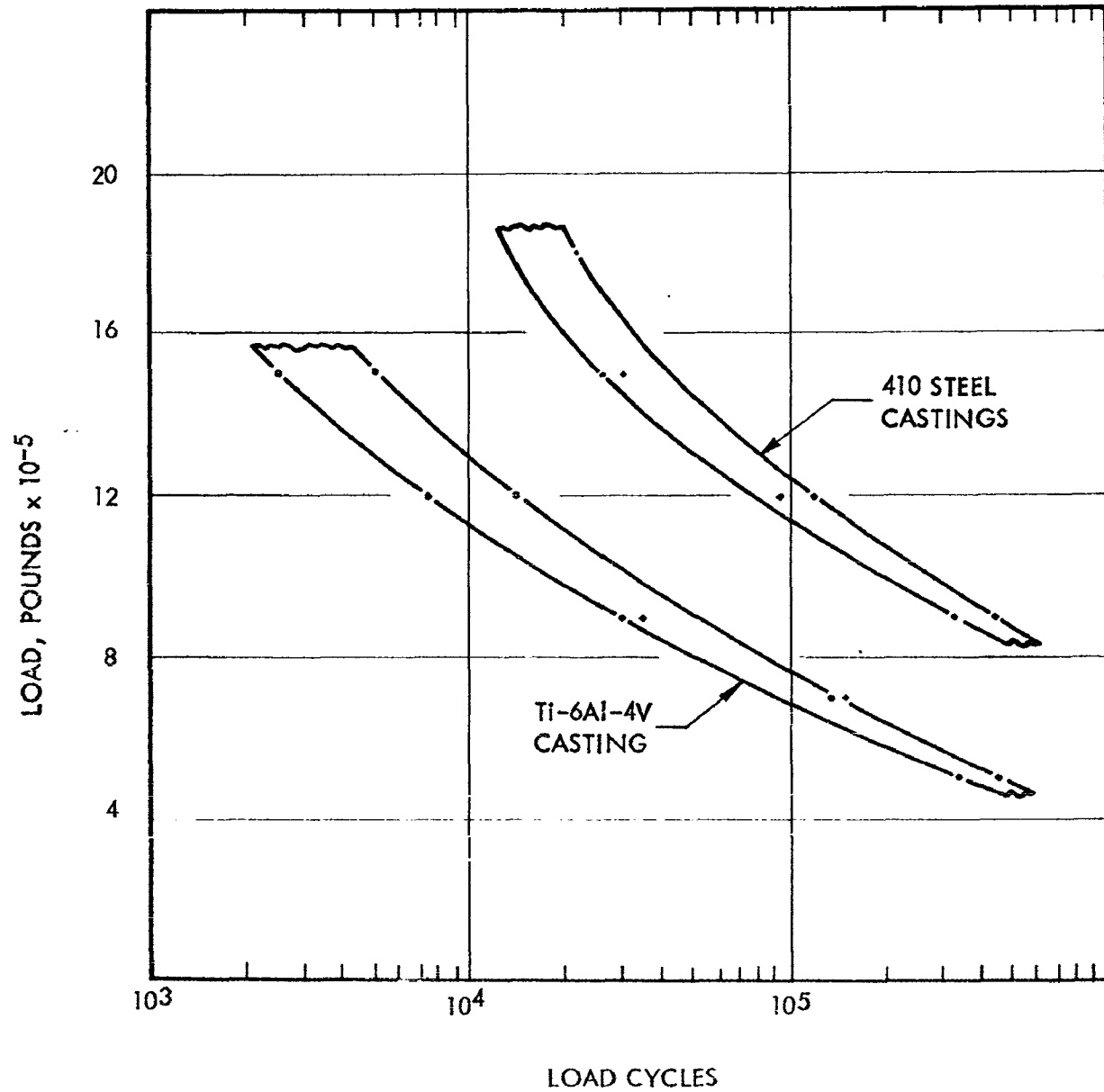


FIGURE E4

RESULTS OF FATIGUE TESTS OF THE DEVELOPMENTAL BRACKET



Testing of The Flap Track Link

Two types of tests were used to evaluate the cast Ti-6Al-4V Flap Track Link. One production AISI 4340 steel part and one cast Ti-6Al-4V part were static compression loaded to failure as shown in Figure E5. Three steel parts and six cast Ti-6Al-4V parts were fatigue loaded to failure.

The AISI 4340 parts were obtained directly from production stock in the finished condition. They had been heat treated to the 180,000 to 200,000 psi ultimate strength range. The Ti-6Al-4V castings were in the as-cast condition and had been pickled to remove .015 inches material thickness per surface. The static tests failed the steel component at 127,600 pounds load and the cast Ti-6Al-4V component failed at 104,700 pounds load. The mechanism of failure was uniform bending of the shank for the steel part and localized crippling of the tubular portion of the cast Ti-6Al-4V part.

The fatigue tests were conducted in a fixture designed to load the part in combined bending and torsion, to simulate expected service conditions. The torsional load on the part is equal to 6.17 times the bending load. The results of the fatigue tests are as follows:

Material	Bending Load, Pounds	Cycles to Failure
Steel	5000	4,667
	3000	24,330
	2000	148,231
Cast Ti-6Al-4V	2000	11,602
	2000	6,067
	1500	66,408
	1500	57,557
	1000	216,833
	1000	1,000,000 (Not failed)

Figure E6 is a S-N curve of these test results.

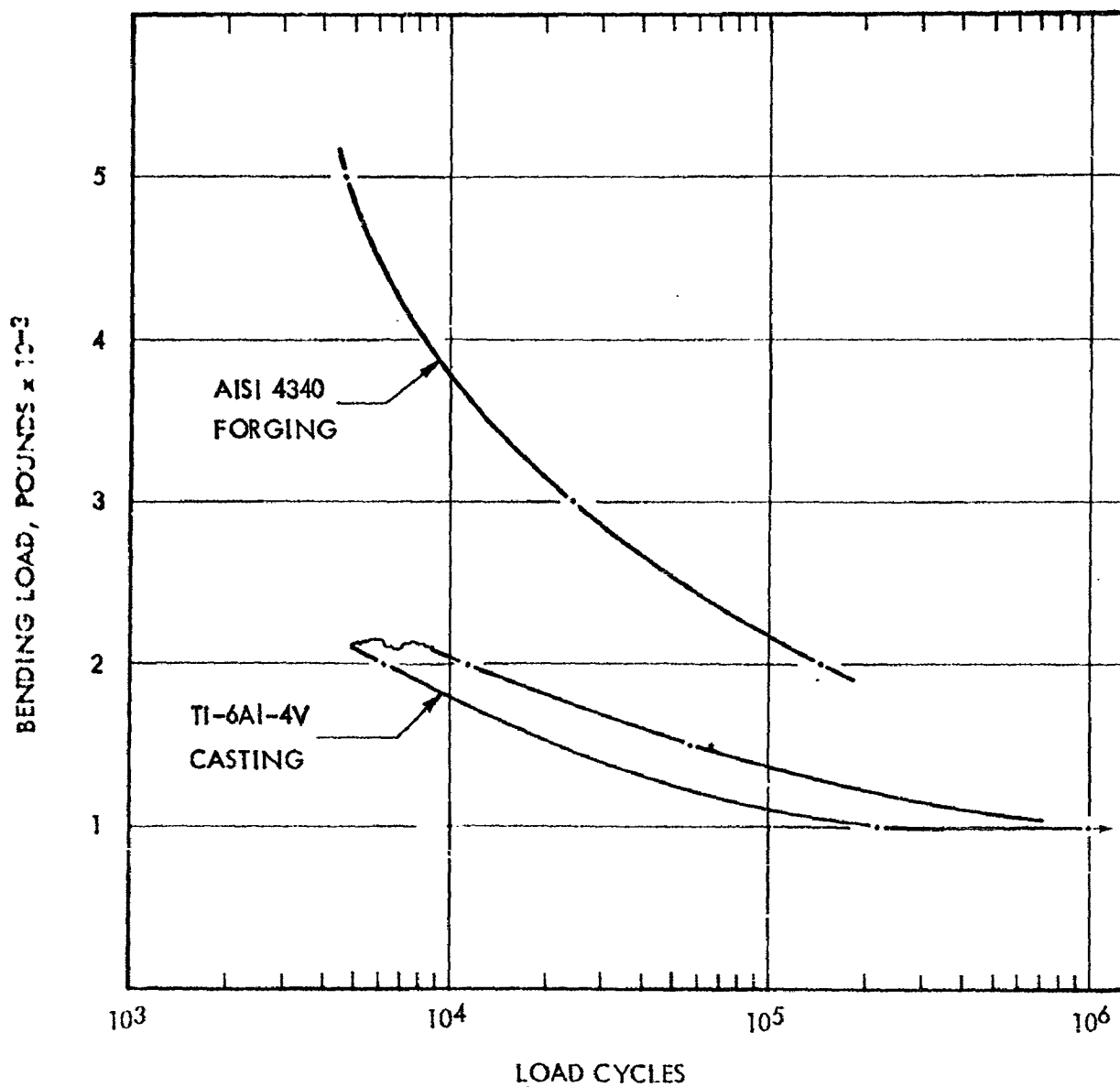
FIGURE E5

FIXTURE FOR COMPRESSION TESTING OF FLAP TRACK LINK



FIGURE E6

RESULTS OF FATIGUE TESTS OF THE FLAP TRACK LINK



Testing of The Sway Brace Case

Static tests were used to compare the properties of five cast Ti-6Al-4V components with two production cast type 410 steel components heat treated to the 150,000 psi ultimate tensile strength level. The Ti-6Al-4V castings were in the as-cast condition.

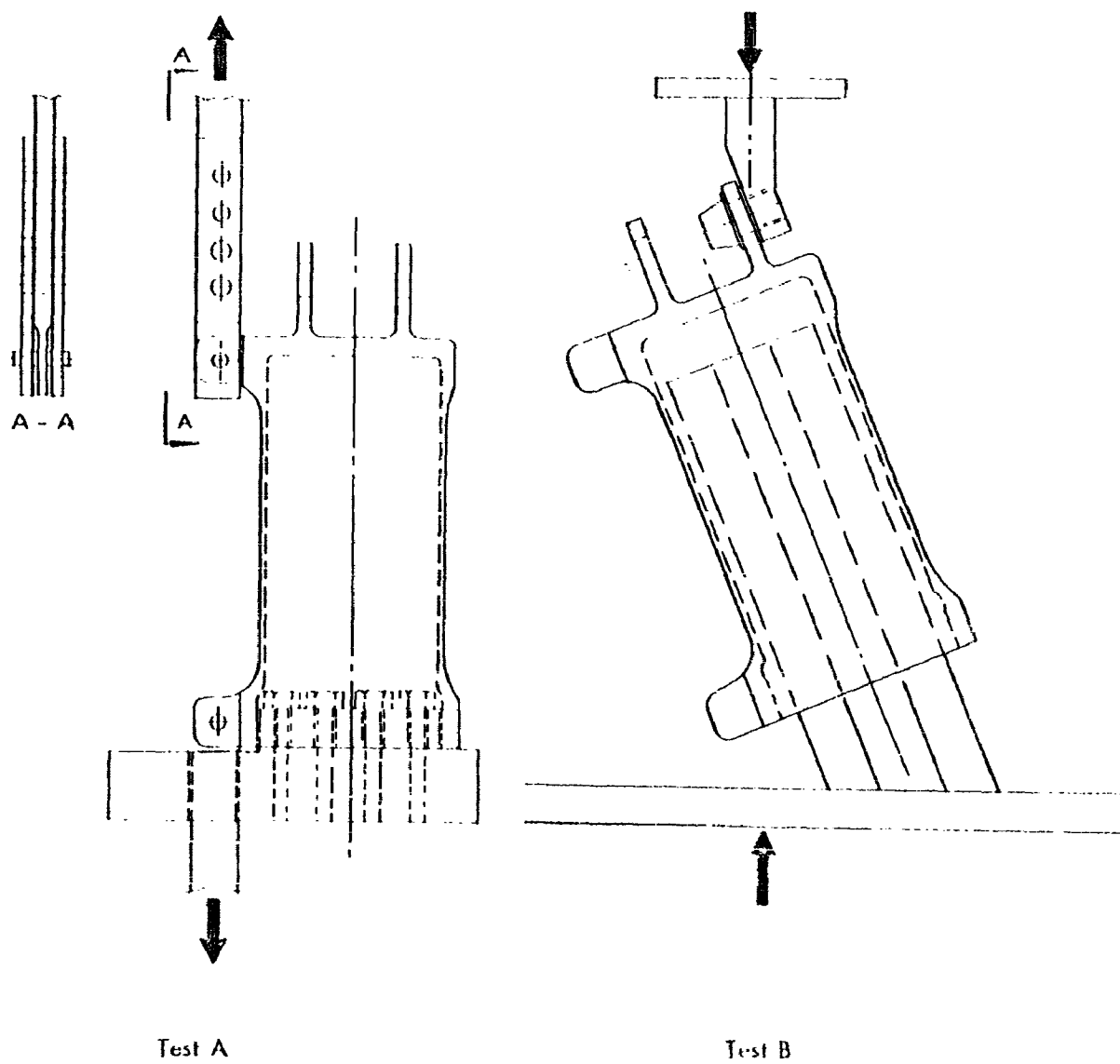
The test setup and loading conditions were as shown in Figure E7 and were designed to simulate actual service loads.

No casting failures were experienced in the tests which loaded the large lugs in compression (Test B in Figure E7). In all trials, the bolt which loaded the lug failed in shear.

The 410 steel cast Sway Brace Cases tested as shown in Test A (Figure E6) failed at 32,700 pounds and 35,300 pounds tension load. Both failures were in the upper lug and exhibited good ductility. The five Ti-6Al-4V castings failed at 22,900; 23,400; 25,000; 25,000; and 28,600 pounds tension loads at the same location. These failures exhibited fair ductility but not equivalent to the 410 steel test parts.

FIGURE E7

TEST FIXTURES FOR THE SWAY BRACE CASE



Testing of The Elevator Torsion Fitting

This component is a part of a missile control system and was designed for rigidity rather than strength or fatigue requirements. Consequently, the evaluation tests were comparisons of rigidity of the cast Ti-6Al-4V part and the production AISI 4340 steel part which was machined from bar stock. The Ti-6Al-4V parts were in the as-cast condition and the steel parts had been heat treated to the 180,000 - 200,000 psi ultimate strength level. (Since these parts were obtained, this production component has been redesigned as a type 410 steel casting.)

The Torsion Fittings were tested in a fixture adapted to a conventional torsion test machine. The torsion load was fed into the slotted center hole and the reaction load restrained at the channel ends. Deflection of the parts was measured by dial indicators supported by the load input bolts and cantilevered to the channel ends. A schematic load diagram is shown in Figure E8. Typical load-deflection curves obtained from these tests are shown as Figures E9 and E10.

FIGURE E8

LOAD SCHEMATIC FOR ELEVATOR TORSION FITTING

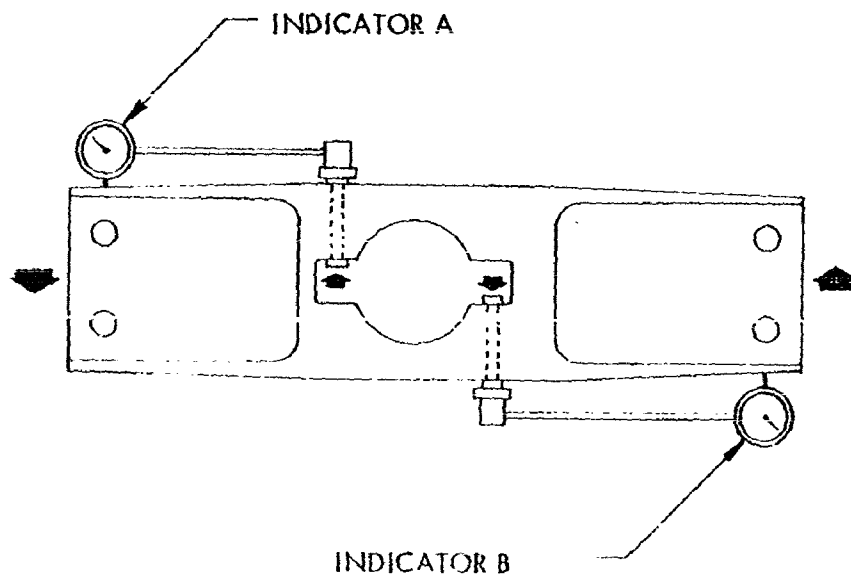


FIGURE E9

RESULTS OF DEFLECTION TEST OF CAST Ti-6Al-4V ELEVATOR TORSION FITTING

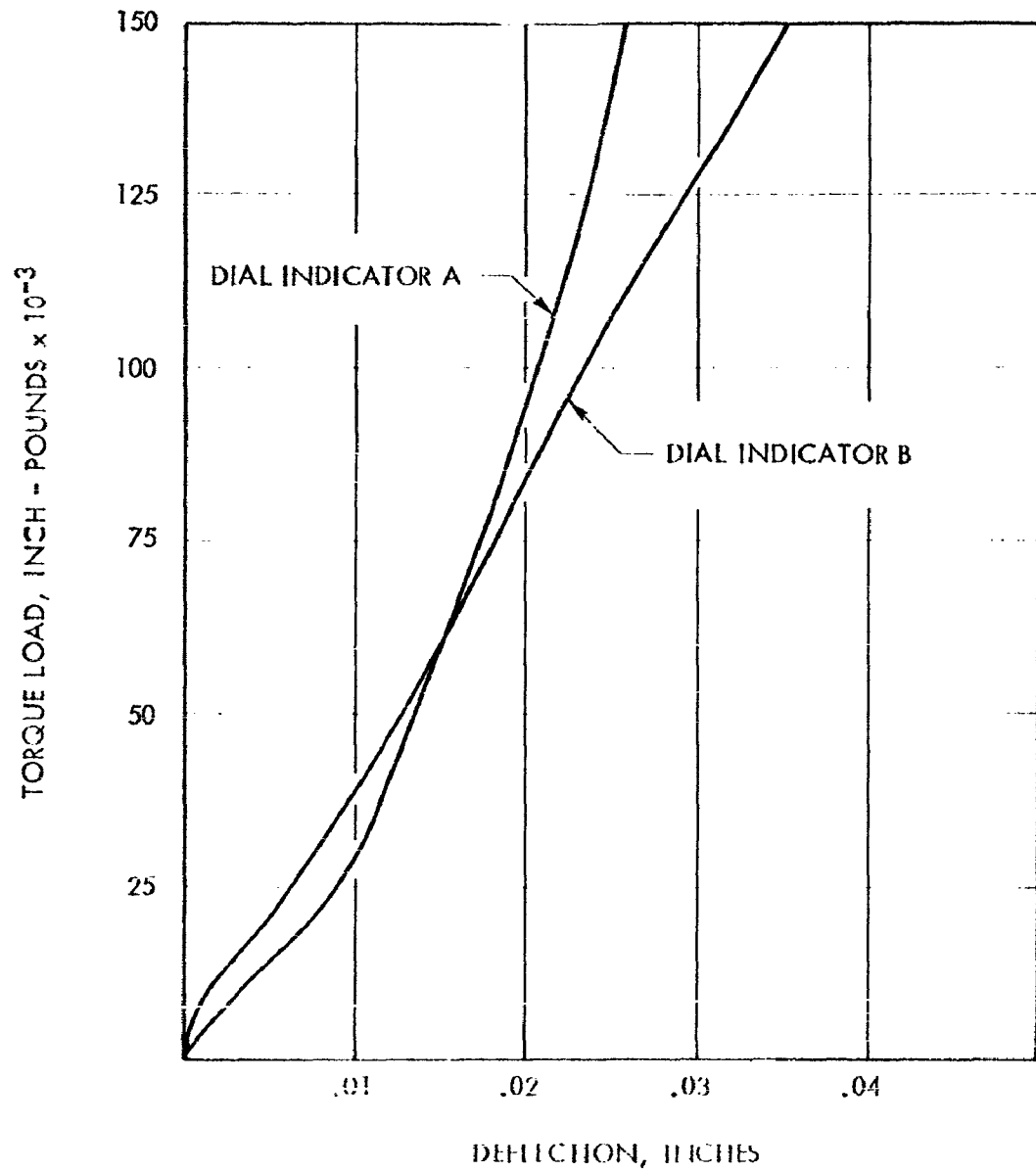
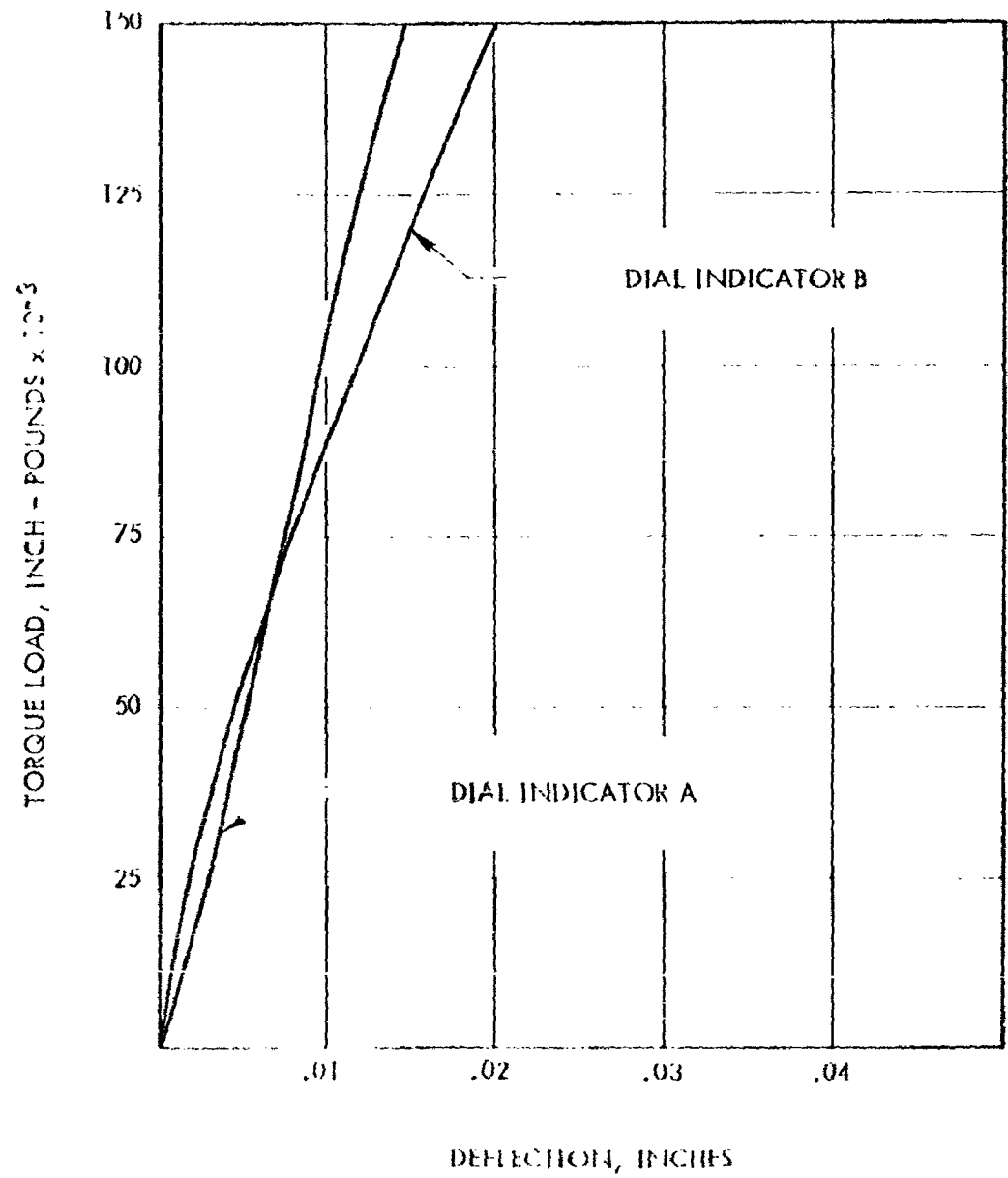


FIGURE F10

RESULTS OF DEFLECTION TEST OF A STEEL ELEVATOR TORSION FITTING



Testing of The Inboard Horizontal Stabilizer Rib

Since this part was not satisfactorily produced as a titanium alloy casting, no tests were conducted.

D2-2786-8

F1

SECTION F

SPECIFICATION AND DESIGN CRITERIA

DEVELOPMENT OF CASTING SPECIFICATIONS

A tentative material specification for procurement of air-frame quality Ti-6Al-4V castings has been developed and is presented on page F3. This specification is based on the conclusions made from the development work conducted in this program and from experience gained in development and use of other casting specifications.

TENTATIVE SPECIFICATION - TITANIUM ALLOY CASTINGS
(Ti-6Al-4V)

Tentative Boeing Material Specification BMS 7-111
Oregon Metallurgical Corporation Specification--OMC 164

1. ACKNOWLEDGMENTS:

- 1.1 A vendor shall mention this specification number in all pertinent quotations and when acknowledging purchase orders.

2. APPLICABLE SPECIFICATIONS:

- 2.1 The following specifications, of the issue in effect on date of invitation for bids, form a part of this specification to the extent herein described:
- 2.1.1 MIL-I-6865 Inspection, Radiographic
 - 2.1.2 MIL-I-6866 Inspection, Penetrant
 - 2.1.3 MIL-C-6021 Castings, Classification and Inspection
 - 2.1.4 Federal Test Method Standard No. 151
 - 2.1.5 Reference Radiographs for Titanium Castings (Boeing Document D2-2786-9)

3. APPLICATION:

- 3.1 This specification is primarily for high strength titanium alloy aircraft castings requiring room temperature ultimate tensile strength above 140,000 psi, light weight, and good corrosion resistance.

4. TYPES AND GRADES:

- 4.1 Unless otherwise specified, castings are to be in the as-cast condition. If annealed condition is specified, the annealing heat treatment shall be 1000°F for 4 hours followed by air cool.
- 4.2 Casting classification shall be as specified on the applicable casting drawing. Class definitions shall be as specified in MIL-C-6021.

5. CASTING METHOD:

- 5.1 Casting molds shall be rammed graphite mixed with suitable binders, unless otherwise specified.

- 5.2 Melting shall be by the vacuum double-melt process when the titanium source is either raw sponge or raw sponge compacts. Use of titanium solids in preparation of electrode stock is permitted, provided all other requirements of this specification are met. All melting shall be by the vacuum consumable-electrode process.

6. COMPOSITION:

- 6.1 Castings shall be of the following composition:

	<u>Weight Per Cent</u>
Carbon	.10 max.
Hydrogen	.015 max.
Nitrogen	.07 max.
Oxygen	.25 max.
Iron	.30 max.
Aluminum	5.5 - 6.5
Vanadium	3.5 - 4.5
Other Elements	.4 max.
Titanium	Balance

- 6.2 Each heat shall be analyzed for each of the elements listed in paragraph 6.1 except titanium. Chemical analysis of each lot shall be certified by the vendor. Analysis for hydrogen shall be on samples removed after all vendor processing. All analyses shall be performed using equipment and procedures approved by the procuring agency. Hydrogen analysis sample shall be removed from a location designated on the casting drawing, or from a separate coupon from the same heat and processed at the same time as the casting lot. This coupon shall be no thicker than the thinnest section of the casting.

7. QUALITY:

- 7.1 Castings shall be uniform in quality and condition, well cleaned, and have a uniformly smooth surface compatible with the casting process. Castings shall be free of surface contamination (such as oxygen, nitrogen, or other foreign contaminants).
- 7.2 Unless otherwise specified, metallic grit or shot shall not be used for final cleaning.
- 7.3 Castings shall not be repaired by plugging, welding, peening, or other methods without written permission of the procuring agency.

- 7.4 The areas of castings subject to soundness requirements shall be as specified, and the number and extent of defects in such areas shall not be greater than indicated by the standard furnished or approved by the procuring agency. Combinations of cracks, shrinkage cavities, cold shuts, misruns, or other defects not individually cause for rejection, but which are so aligned as to cause stress concentration are cause for rejection.
- 7.5 When soundness is specified in accordance with paragraph 7.4, it shall be determined in accordance with visual, penetrant, and radiographic inspection methods established by MIL-C-6021, MIL-I-6865, and MIL-I-6866.

8. TENSILE PROPERTIES:

- 8.1 All test specimens shall be cast in a graphite mold in the same heat and in the same manner as the castings which they represent.
- 8.2 Two cast test bars per heat shall be furnished to the purchaser. Bars shall be of sufficient size to be machined into type R3 specimen in accordance with Method 211.1 of Federal Test Method Standard No. 151.
- 8.3 Vendors shall conduct at least two tensile tests per heat.
- 8.4 Tensile tests are to be performed in accordance with Federal Test Method Standard No. 151 using type R3 specimen. Strain rate shall be $.005 \pm .002$ in/in/min through 0.2 per cent strain.
- 8.5 Tensile properties of each separately cast specimen or specimen sectioned from a critical area of a 1A casting shall meet or exceed the following minimum values:

Ultimate tensile strength - - - - -	140,000 psi
Yield tensile strength (.2% offset) - - - - -	122,000 psi
Elongation - - - - -	5 per cent
Reduction of area - - - - -	10 per cent

9. IDENTIFICATION:

- 9.1 Unless otherwise specified, each casting shall be identified with the part number and a vendor identification symbol approved by the purchaser by the use of raised figures in a location indicated on the drawing. When no location is shown on the drawing, the number shall be so located as not to be machined off in finishing to the required casting dimensions. Such numbers shall not be at indicated tool point locations.

10. CHEMICAL REMOVAL OF SURFACE MATERIAL

- 10.1 If pickling is specified, castings shall be pickled to remove .015 inch minimum material thickness per surface. Foreign material, such as particles of mold, adhering to the casting surface shall be removed prior to pickling. Castings shall not be peened, abrasive blasted or otherwise finished after pickling, unless otherwise specified.

11. CERTIFICATION:

- 11.1 Three copies of a quality certification shall accompany or precede each lot of castings. The certification shall include results of analyses in accordance with paragraph 6.2, the results of tensile tests in accordance with paragraphs 8.3 and 8.4, the vendors heat number, the thickness of material removed by pickling in accordance with paragraph 10.1, the quantity of castings constituting that lot, the casting part number, and the purchase order number.

12. DEFINITIONS:

- 12.1 A lot consists of castings of the same heat, the same configuration, the same condition, processed at the same time, and submitted for inspection at the same time.
- 12.2 A heat consists of the material produced in one melting and pouring cycle.

DEVELOPMENT OF MINIMUM PROPERTY VALUES

A group of 196 heats of cast Ti-6Al-4V alloy were studied to establish minimum property values for separately cast test bars, for design purposes. The heats were all those produced prior to the production run which were within the composition limits established by the specification (page F3) and for which mechanical properties data were available, with the exception of a small number of experimental low-interstitial heats which were poured during study of melting practice.

The ultimate tensile strength, yield strength, elongation, and reduction of area values for these heats were averaged and the standard deviations determined by conventional statistical analysis methods as shown in the appendix to this report.

Both "A" and "B" design values were determined. For a "normal" distribution, "A" values assure to a 95% confidence level that 99 percent of the distribution will exceed the "A" value given. The "B" design value assures to the same confidence level that 90 per cent of the distribution will exceed the "B" value. Although mechanical property distributions are not "normal" in that the distribution curve is slightly skewed toward the high-value end of the curve, this analysis method is convenient and the small error involved is in the conservative direction.

In accordance with Convair Astronautics Document AZS-27-274 "Statistical Determination of Strength Properties", 2.573 standard deviations were subtracted from the property averages to establish "A" design values and 1.452 standard deviations were subtracted from the averages to establish "B" values. This method is usually applied to strength values, but is also suitable for establishing the ductility minimums. The averages, "A" values, and "B" values obtained were as follows:

	Average	"A" Minimum	"B" Minimum
Ultimate Tensile Strength, KSI	146.8	135.0	140.0
Yield Strength, KSI, .2% Offset	129.8	116.0	122.0
Elongation, per cent	8.0	3.2	5.3
Reduction of Area, per cent	15.3	5.7	9.6

COMPRESSION, BEARING, AND SHEAR PROPERTIES

Room temperature compression, bearing, and shear tests were conducted to provide basic properties values for design purposes. Tensile tests from the same heats were also conducted for reference. The results were as follows:

Tensile Tests

Heat Number	Ultimate Tensile Strength, KSI	Yield Strength, KSI (.2%)	Elong. Per Cent	R.A., Per Cent
P441	152.5	138.0	9	17
P483	141.3	129.0	12	18
P494	143.5	130.5	10	17

Compression Tests

Heat Number	Yield Strength, KSI (.2%)	Modulus of Elasticity, KSI
P441	141.0	18.3
P483	148.7	17.9
P494	135.5	17.7

Bearing Tests

Heat Number	Ultimate Bearing Strength, KSI	Yield Bearing Strength, KSI*	Hole Diameter, Inches	Edge Margin, Inches
P441	239.1	215.9	.1599	.2566
P483	225.0	194.1	.1600	.2573
P494	235.0	208.9	.1599	.2574

* Based on offset of 2 per cent of hole diameter

Shear Tests

Heat Number	Ultimate Shear Stress, KSI
P441	96.3
P483	90.8
P494	97.3

DEVELOPMENT OF REFERENCE RADIOGRAPHS

X-ray reference radiographs were prepared from specimens machined from the discs which were cast during the feeding studies. The radiographs of the discs of proper thickness (1/4, 1/2, and 1 inch thicknesses) were examined and sections which represented reasonable gas and shrinkage porosity quality gradients were selected and machined from the discs. Other casting imperfections found in steel, aluminum, and magnesium alloy castings (such as heavy inclusions, dross, micro-shrinkage, segregation, internal cracks, and internal cold shuts) were not experienced in the titanium alloy castings, with the exception of mold material (graphite) inclusions. This defect was not often observed, and when it was found, it was similar in appearance on the radiograph to shrinkage porosity. Consequently, the defects shown in the shrinkage porosity reference radiographs can be applied to graphite inclusions, since the appearance and stress concentration effects of these types of defects are similar. The reference radiographs are attached to the back cover of this report so that they may be readily removed for use. These radiographs were prepared to a film density of 1.9 to 2.2 (American Standard Printing Density P 2-3) so that they may be directly compared to typical casting radiographs.

DEVELOPMENT OF DIMENSIONAL TOLERANCE CRITERIA

Nine dimensions were selected from the Development Bracket and the Flap Track Link for analysis of deviations from the desired dimensions. The dimensions were selected to enable separation of the effects of parting plane causes of deviation from the basic tolerance characteristics of the rammed graphite mold casting process. Fifty of each part were studied.

There are several obvious sources of dimensional variance in castings. Examples are; deviation of the pattern dimensions from the desired nominal dimensions, the inaccuracy of the established average shrinkage of mold and metal, the variations in mold and metal shrinkage from the established average, the inconsistencies in mold ramming and drawing practice from mold to mold, the inability of the foundry to establish the proper pattern dimensional corrections after a casting trial (due to the impracticality of pouring enough trial casting to establish true process average dimensions), variations in casting cleanup practice (removal of gates, risers, flash, etc.), variations in pouring temperature and practice, and variations in mold material quality and in mixture ratios.

The effects of these causes of dimensional variation are cumulative and each cause represents a separate avenue for potential improvement in dimensional quality.

The dimensions selected for study represent three general types of tolerances. Four simple dimensions which were not across the parting plane were selected for establishment of a basic tolerance. These dimensions varied from .2 to 9.6 inches. Three dimensions of similar magnitude but across and parallel to the parting plane were selected to separate the effects of pattern and mold mismatch. Two dimensions which were across and perpendicular to the parting plane were studied to separate the effects of poor parting plane fitup other than mismatch.

The results of these studies are in Table J24 and frequency distribution bar charts for each of the dimensions studied are presented as Figures F1 through F9. Recommended minimum design tolerances were computed from these data. The actual values for each dimension were averaged and the standard deviation was then calculated as explained in the Appendix. Three standard deviations were used to

establish the recommended minimum tolerances. Such a tolerance will make approximately 99.7 per cent of the dimensions in a normal distribution acceptable, provided that the average of the distribution is exactly the desired nominal dimension. Obviously, it never will be; therefore, the actual number of rejected dimensions will be greater than the 0.3 per cent implied, the amount dependent on how close the foundry can make their average approach the desired nominal dimensions. The usual practice is to cast one or two trial pieces, inspect them, and make pattern corrections. This practice will reveal only large errors, since any dimension on one individual part will rarely indicate the average of a large number of pieces.

General rules for recommended design dimensional tolerances for dimensions up to ten inches were derived from the results of this study, as follows:

- (a) For simple dimensions under one inch that do not cross the mold parting plane, use a basic dimensional tolerance of plus or minus .015 inch. If the dimension exceeds one inch, add .005 inches to the basic tolerance for each whole inch. Examples are $0.620 \pm .015$, $1.135 \pm .020$, $4.720 \pm .035$.
- (b) For dimensions that cross the parting plane and are essentially perpendicular to it, add .008 inch to the tolerance calculated as in (a) above. Examples are $0.620 \pm .023$, $1.135 \pm .028$, $4.720 \pm .043$.
- (c) For dimensions that cross the parting plane and are essentially parallel to it (i.e. mismatch), add .020 inch to the tolerance calculated as in (a) above. Examples are $.620 \pm .035$, $1.135 \pm .040$, $4.720 \pm .055$.

These dimensional tolerances represent minimums. Larger tolerances should be used when practical. It is anticipated that some improvement in available minimum tolerances (particularly in mismatch) will occur as the mold preparation process becomes mechanized. All rammed molds in this program were manually prepared with the aid of a pneumatic rammer.

FIGURE F1

DIMENSION DISTRIBUT IN DEVELOPMENTAL BRACKET

This dimension is basic, both surfaces being in the same mold half. The desired dimension was $.200 \pm .015$ inches.

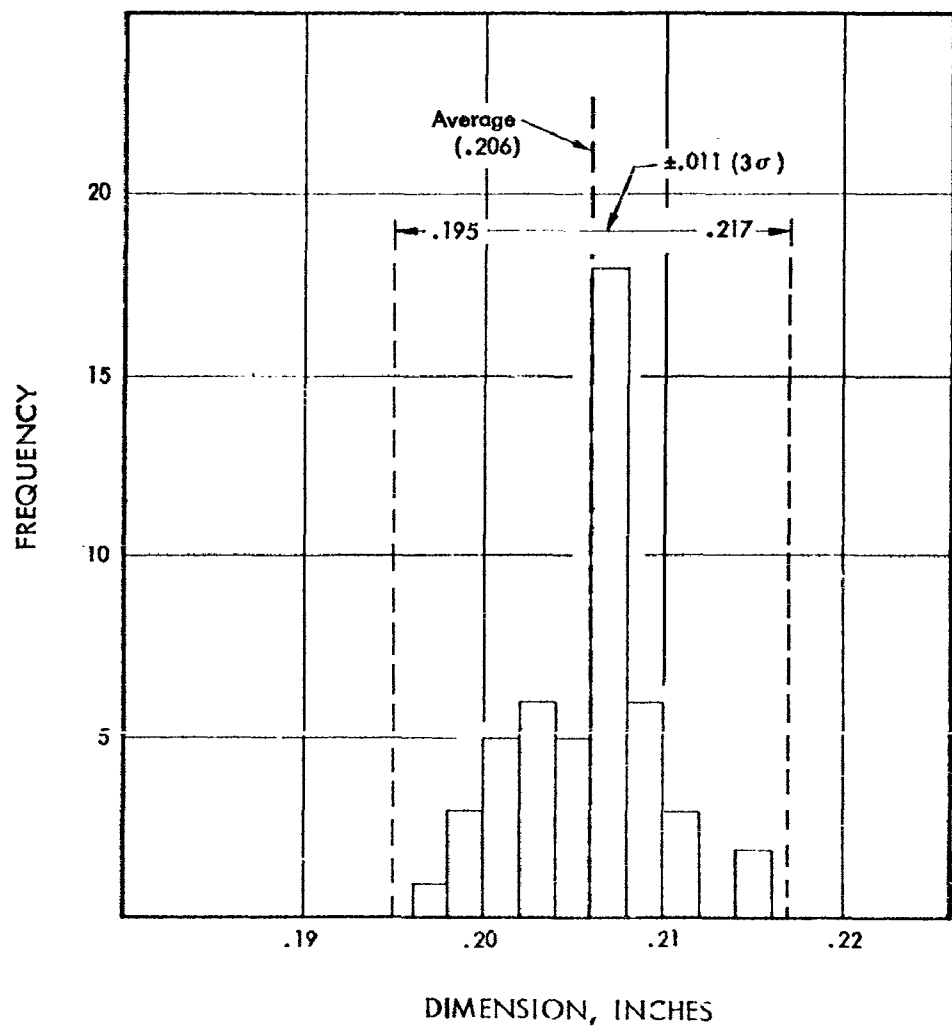


FIGURE F2

DIMENSION DISTRIBUTION IN DEVELOPMENTAL BRACKET

This dimension is basic, both surfaces being in the same mold half. The desired dimension was $.650 \pm .015$ inches.

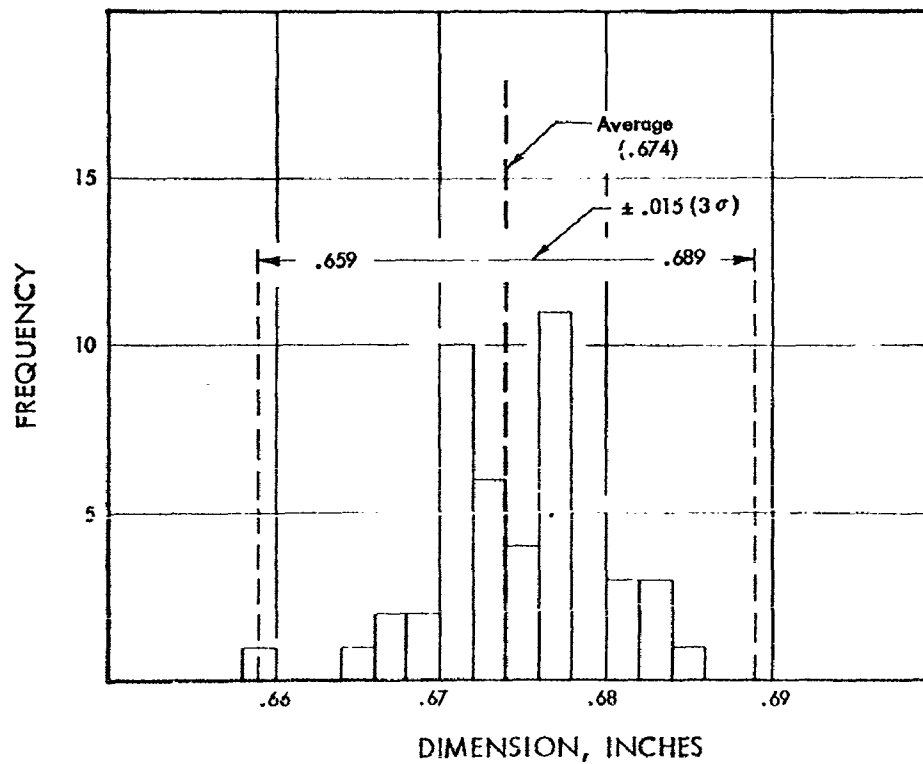


FIGURE F3

DIMENSIONAL DISTRIBUTION IN DEVELOPMENTAL BRACKET

This dimension is basic, both surfaces being in the same mold half. The desired dimension was $3.050 \pm .015$ inches.

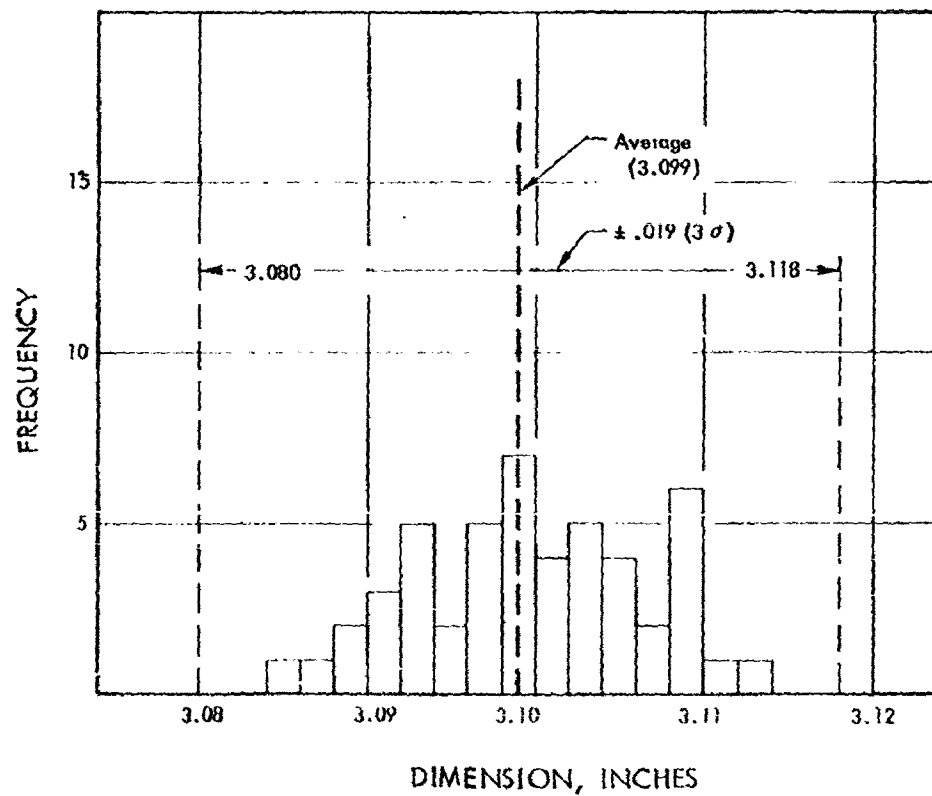


FIGURE F4

DIMENSIONAL DISTRIBUTION IN FLAP TRACK LINK

This dimension is basic, both surfaces being in the same mold half. This dimension was not directly specified on the engineering drawing, but is controlled by two cumulative dimensions. It was selected because it was convenient to measure and represents a large dimension on the part.

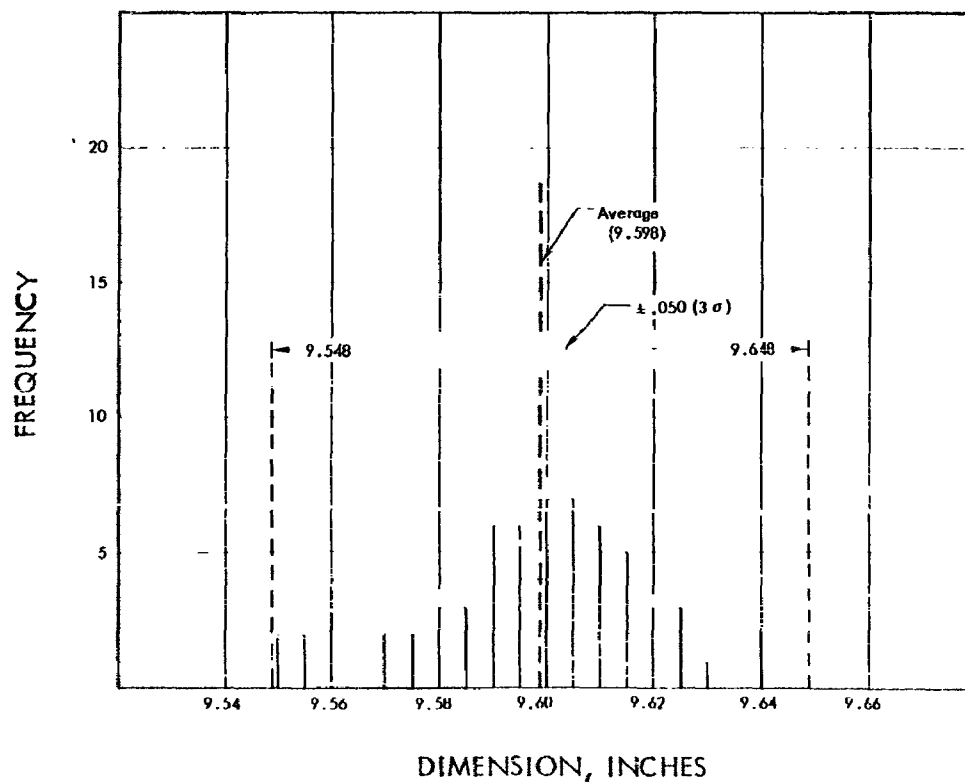


FIGURE F5

DIMENSIONAL DISTRIBUTION OF DEVELOPMENTAL BRACKET

This dimension is across the parting plane and is perpendicular to it. It represents the basic tolerance plus the added dimensional variations resulting from misfit of the mold parting plane. It does not involve mismatch. The desired dimension was $1.520 + .025 - .015$ Inches.

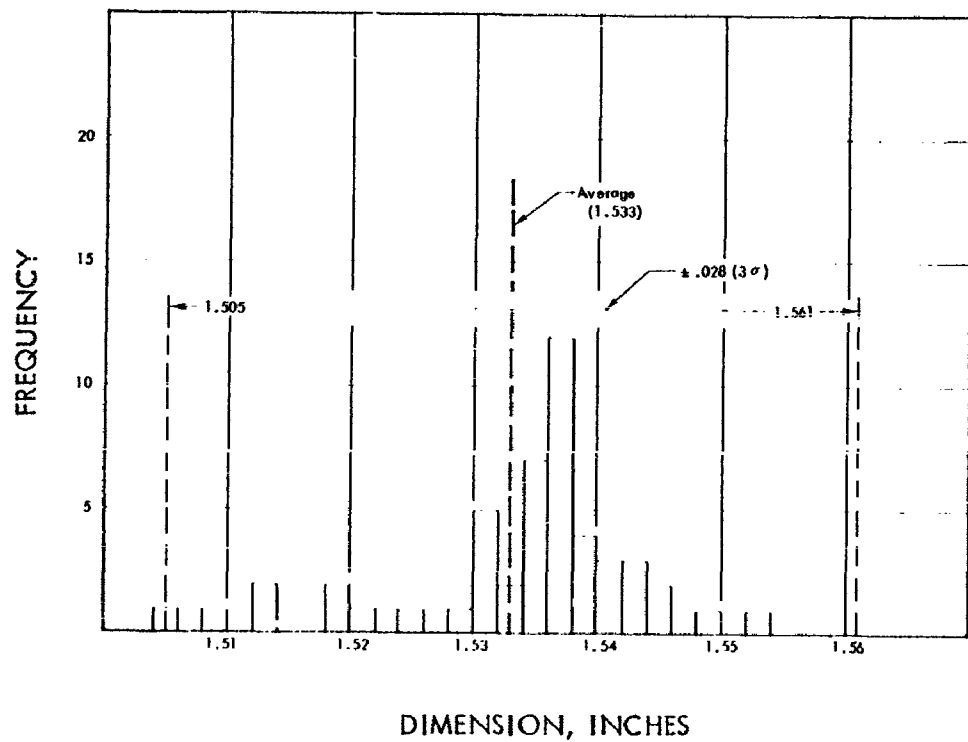


FIGURE F6

DIMENSIONAL DISTRIBUTION OF DEVELOPMENTAL BRACKET

This dimension is across the parting plane and is perpendicular to it. It represents the basic tolerance plus the added dimensional variations resulting from misfit of the mold parting plane. It does not involve mismatch. The desired dimension was $2.520 \pm .025 - .015$ inches.

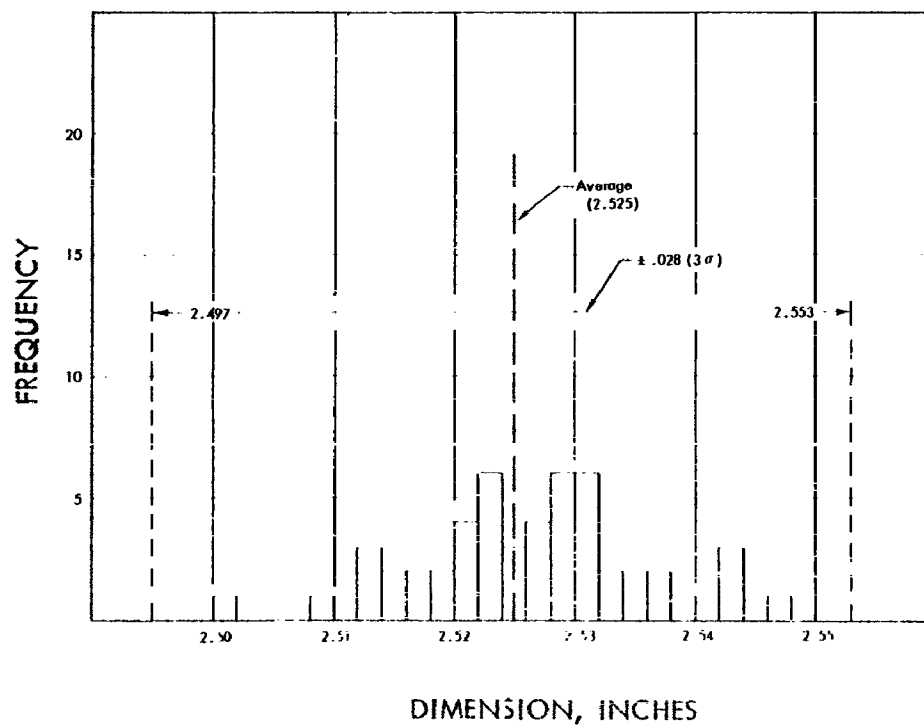


FIGURE F7

DIMENSIONAL DISTRIBUTION OF DEVELOPMENTAL BRACKET

This dimension is across the parting plane and is parallel to it. It represents the basic dimensional tolerance plus the added variations resulting from pattern and mold mismatch. The desired dimension was not directly specified. It was measured for comparison with Figure F2 to separate mismatch effects.

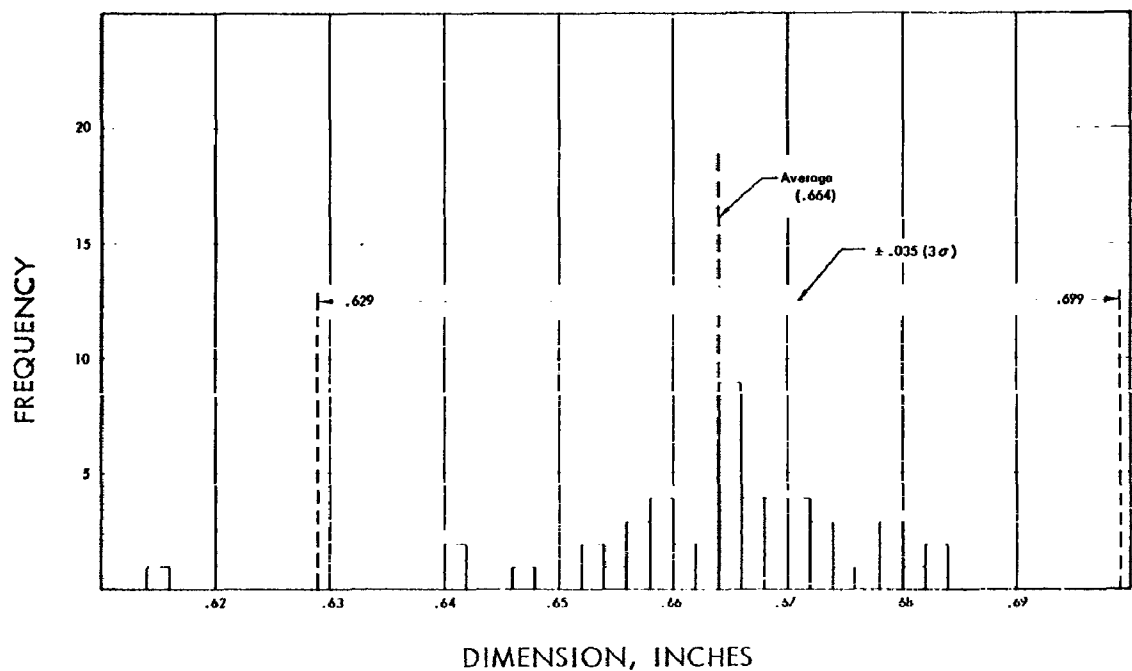


FIGURE F8

DIMENSIONAL DISTRIBUTION OF DEVELOPMENTAL BRACKET

This dimension is across and parallel to the parting plane. It represents the basic tolerance plus effects of pattern and mold mismatch. The desired dimension was $3.050 \pm .015$ inches.

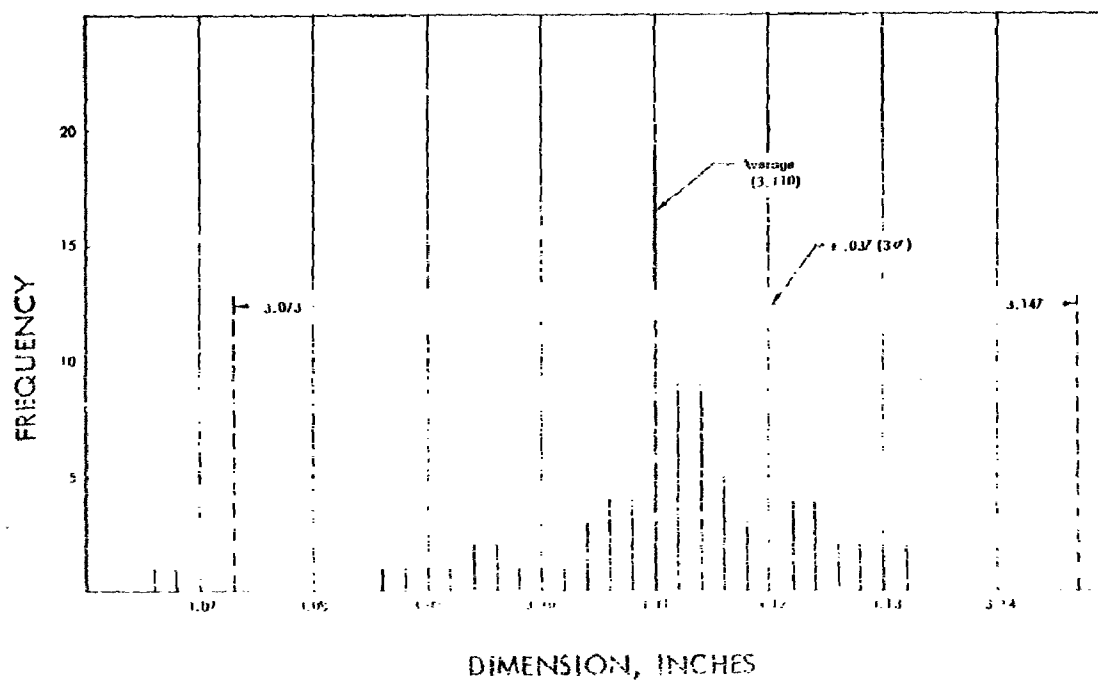
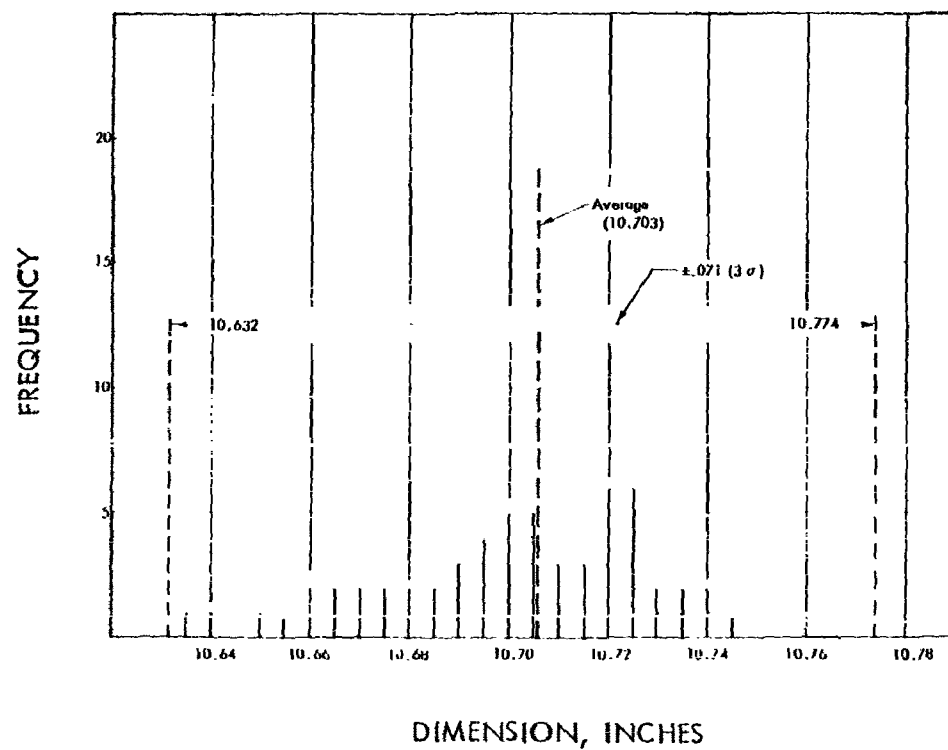


FIGURE F9

DIMENSIONAL DISTRIBUTION OF FLAP TRACK LINK

This dimension is across and parallel to the parting plane. It represents the basic tolerance plus effects of pattern and mold mismatch. The desired dimension was $10.630 \pm .050$ inches.



D2-2786-8

G1

CONCLUSIONS

SECTION G

The following conclusions are based on the development work performed during this program. They are listed in the same order as the corresponding discussions in this report, except for the first three conclusions which are of a general nature.

1. A complete commercially feasible process was developed for production of titanium alloy castings of the type, sizes, and quality required for aircraft and missile application.
2. A practical demonstration of the process was accomplished by production of quantity lots of several airframe structural shapes.
3. A procurement specification, quality control procedure, inspection standards, and design criteria were established to permit production application of titanium alloy castings.
4. A vacuum, consumable-electrode, arc furnace with water-cooled copper, tilt pour crucible is suitable for melting and pouring of production heats of titanium alloys.
5. Preparation of melting electrode stock may be satisfactorily done by inert-gas shielded arc welding of pressure-compacted titanium sponge mechanically mixed with the proper alloying additions, or by similar welding of titanium solids from previous melting operations. Care in welding must be exercised to minimize oxygen and nitrogen contamination of the electrode stock.
6. Foundry and other scrap may be satisfactorily recycled to produce aircraft quality titanium castings, provided additional low-interstitial stock is added to dilute the interstitial content. It will be necessary to occasionally discard foundry scrap if the interstitial level becomes too high. This requirement can be minimized if electrode welding is conducted in an inert gas chamber.
7. An expendable rammed graphite mold has been found suitable for production casting of titanium alloys. Properties of the mold were developed and are presented and discussed in Section B of this report.

8. A graphite-base shell mold was developed, which with further development aimed at increasing its resistance to penetration, offers considerable potential as a production mold. At the present stage of development, shell graphite molds are suitable for small shapes.
9. Development of an investment graphite mold suitable for casting of titanium was partially accomplished. The successful development of such a mold appears feasible if based on the use of graphite.
10. Used rammed graphite mold materials were satisfactorily reused to make new molds.
11. Feeding characteristics were determined for titanium alloy Ti-6Al-4V cast in rammed and machined graphite molds.
12. Shrinkage porosity in cast titanium appears as distinct voids on the X-ray film, rather than as cloudy low-density areas.
13. Less taper is generally required in rammed than in machined graphite molds, for equivalent soundness.
14. Increasing the amount of taper progressively improves feeding distance in rammed molds. In the case of machined graphite molds the feeding distance decreased until taper exceeded three or four degrees, and then increased.
15. The effectiveness of taper decreases as the section thickness is increased.
16. The tapers required to cast sound sections were established in relation to thickness and are shown in Figure B18.
17. The use of heated molds did not appreciably improve the feeding of cast Ti-6Al-4V but did decrease the gas porosity problem, apparently because of reduced moisture pickup during mold assembly.
18. The feeding distance in shell graphite molds appeared to be significantly superior to rammed and machined graphite molds.

19. The feeding characteristics of centrifuge cast Ti-6Al-4V discs were not significantly different from the similar discs which were statically cast.
20. Castings which are centrifuge cast should be fed such that the metal enters the mold from the trailing side of the casting and from the outside (away from the centrifuge axis). This is necessary to minimize poor casting surface resulting from turbulent metal flow into the mold cavity.
21. Removal of gates and risers from the castings by power saw or abrasive cut-off wheel have been found to be satisfactory. Use of oxy-acetylene torch increases interstitial contamination of the scrap and requires excessive hand grinding to clean up the casting.
22. Grinding and belt sanding is satisfactory for primary clean up of titanium castings. Grit blasting is satisfactory for removal of mold particles and surface oxides. Sand blasting is satisfactory for final clean up.
23. A satisfactory acid solution and processing technique was developed for chemical removal of surface material from titanium castings.
24. For best fatigue properties, 0.015 inch per surface should be chemically removed from Ti-6Al-4V castings, using the solution and process developed.
25. At the present stage of development, unalloyed titanium and Ti-6Al-4V are suitable for "standard" casting alloys.
26. A satisfactory heat treatment to improve the properties of cast Ti-6Al-4V was not developed, the best properties being obtained in the as-cast condition.
27. Ti-2Cu casting alloy has ductility superior to cast unalloyed titanium which has been strengthened to the same level by interstitial additions.

28. The properties of a titanium casting alloy containing four to six per cent aluminum, four to six per cent tin, eight to ten per cent zirconium, and two to four per cent vanadium are slightly better than those of cast Ti-6Al-4V. Additional work is needed to establish the best composition for this alloy.
29. The effects of the interstitial and alloying elements in cast Ti-6Al-4V were determined. Composition limits were established.
30. Elevated temperature tensile properties for as-cast Ti-6Al-4V were determined.
31. Four of the five shapes selected for study as titanium alloy castings were successfully developed and produced. The fifth shape was too large for the available equipment.
32. Properties of the four developed shapes were compared with the corresponding production components by structural testing.
33. Statistically valid minimum design allowables were established for the as-cast Ti-6Al-4V alloy.
34. Minimum dimensional tolerances for design purposes were determined.

D2-2786-8

H 1

SECTION H

RECOMMENDATIONS

The following recommendations are based on the work done during this contract.

1. Further development of an investment casting mold and process is recommended.
2. Further development of casting alloys is recommended, aimed at higher strength properties.
3. Development of composition limits and a casting specification for the four to six per cent aluminum, eight to ten per cent zirconium, and two to four per cent vanadium titanium casting alloy is recommended, to take advantage of the superiority of this alloy over the Ti-6Al-4V casting alloy.

D2-2786-8

J1

SECTION J

TABLES

TABLES

TABLE J1
MELTING RECORD

Heat No.	Order No.	Material	MELTING			VACUUM - HIGGONS			SKULL WT.		Fluidity Spinn In.	Gross Heat Wt.		Net Heat Wt.	Percent Humidity Yield	
			Time Min.	Elect. Dia.	Cruc. Dia. in.	Start	During Melt	After Pour	5 Min. Leak Rate	Start	Finish	Heat Wt. Lb.	Heat Wt. Lb.	Heat Wt. Lb.	Yield	Percent
P-1	22-1	Ti6Al-4V	3.05	6	9	44		175		6.56		52.25	26.81	51		
P-2	22-1	Ti6Al-4V	4.5	5-6	9	22	170	250		13.75		81.65	27.81	33.5		
P-3	22-1	Ti6Al-4V	3.2	4-5	9	50	125	50		19.0		70.75	29.69	42		
P-4	22-1	Ti6Al-4V	5.0	7	9	37	125	70		15.25		92.75	31.13	34		
P-5	22-2	Ti6Al-4V	6.25	4-5	9	100	175	300		21.25	22.063	99.50	21.19	21		
P-6	22-2	Ti6Al-4V	5.00	4-5	9	105	200	400		30.56		86.31	13.19	15		
P-7	22-1	Ti6Al-4V	5.6	4-5	9	40	200	250		13.61	15.75	97.94	25.19	25.5		
P-8	22-1	Reject-Fourth Short														
P-9	22-1	Ti6Al-4V	6.7	4	9	45	125	200		16.0		92.063	21.0	22.8		
P-10	22-1	Ti6Al-4V	5.6	4	9	50	125	200		35.44		92.063	18.69	20.4		
P-11	22-1	Ti6Al-4V	4.8	3-7	9	50	175	275		27.75	16.31	18.25	90.44	22.063	24.4	
P-12	22-1	Ti6Al-4V	4.9	4-5	9	40	50	175		33.31	15.0	14.5	86.13	16.62	19.3	
P-13	22-1	Ti6Al-4V	5.0	4-7	9	37	180	180		28.50	17.94	16.75	96.50	18.50	19.2	
P-14	22-1	Ti6Al-4V	5.0	4-9	12	100	125	250		37.25	17.88	18.25	118.19	15.063	12.75	
P-15	22-1	Ti6Al-4V	5.35	4-7	9	50	60	200		20.75		18.38	91.37	18.0	19.7	
P-16	22-1	Ti6Al-4V	5.35	4-10	12	60	40	250		21.50	14.0	16	94.56	18.13	19.2	
P-17	22-1	Ti6Al-4V	5.35	4-7	9	60	70	250		22.37	16.25	16	93.37	16.0	17.1	
P-18	22-2	Ti6Al-4V	5.7	4	9	200	200	500		16.25	14.13	13	93.62	13.81	14.8	
P-19	22-5	Ti6Al-4V	4.97	4-5	9	200	300	525		24.13	14.0	Reject	57.0	19.88	20.5	
P-20	22-5	Ti6Al-4V	5.0	4-5	9	50	175	225		30.0	14.0	13	92.56	14.25	15.4	
P-21	22-1	Ti6Al-4V	3.5	3-7	9	80	200	250		21.25	13.50	130.25	39.37	30.3		
P-22	22-7	Ti6Al-4V	5.9	7	9	70	400	100		15.19			9.94			
P-23	22-5	Ti6Al-4V	5.0	4-7	9	70	200	350		21.50	18.62	106.063	15.25	14.5		
P-24	22-7	Reject														
P-25	22-5	Ti6Al-4V	6.0	6-7	12	150	300	750		40.69	24.69	16	127.75	21.75	16.5	
P-26	22-6	Ti6Al-23Sn	5.4	9	9	100	80	200		33.69	22.31	16	101.37	33.81	33.5	
P-27	22-5	Ti6Al-4V	4.75		9	70	100	300		38.13	17.0	18.25	96.37	15.37	16.0	
P-28	22-1	Ti6Al-4V	2.9		9	45	175	225		30.88	24.62	None	51.0	28.62	56.5	

TABLES

TABLE J1 (Continued)

MELTING RECORD

Heat No.	Order No.	Material	MELTING			VACUUM - MICRONS			SKILL WT.		Fluidity Spiral In.	Gross Heat Wt. Lb.		Net Heat Wt. Lb.		Percent Humidity Yield %
			Time Min.	Elect. Dia.	Crucible Dia. In.	Start	Stop	After 5 Min. Pour	Start Lb.	Finish Lb.						
P-29	22-2	Ti6Al-4V	5.4		5	275		750	30.0	12.75	14	66.50		11.69		3.5
P-30	22-1	Ti6Al-4V	4.7		5	70		300	25.13	13.50		65.0		13.81		21.4
P-31	22-4	Ti6Al-4V	3.7	6	9	50		175	17.75	13.50	16	64.19		9.50		15.3
P-32	22-1	Ti6Al-4V	4.6	4-6	9	45		250	33.13	14.37	13.5	93.66		23.62		22
P-33	22-6	Ti5Al-2.5Sn	5.0	4-7	5	70		175	36.81	15.0		93.88		32.44		34.6
P-34	22-5	Ti6Al-4V	4.5	6	9	70		200	30.50	16.50	14	92.0		15.44		17.9
P-35	22-2	Ti6Al-4V	4.35	6	9	70		350	30.0	15.44	12	83.50		11.66		14.25
P-36	22-5	Ti6Al-4V	5.6	6-9	12	250		750	40.0	29.25	Misrun	113.62		20.62		17.7
P-37	22-2	Ti6Al-4V	4.8		5	150		500	34.37	16.25	13	100.37		16.56		16
P-38	22-12	Ti-13V-11Cr-2.5Al	5.3	7	5	40		150	6.19	13.19	10	80.37		30.19		37.6
P-39	22-1	Ti6Al-4V	3.2		9	100		250	(Turnings) 34.0	13.0	None	74.25		42.13		56.7
P-40	22-2	Ti6Al-4V	5.27		9	150		500	31.44	17.0	10	90.37		16.66		18.7
P-41	22-4	Ti6Al-4V	3.52		5	70		275	26.13	10.66	16	75.50		15.50		20.5
P-42	22-2	Ti6Al-4V	5.5	6	9	300		600	25.75	16.75	10	103.50		17.44		16.5
P-43	22-2	Ti6Al-4V	3.75		12	65		250	33.31	26.6	12	70.37		10.13		14.4
P-44	22-12	Ti-13V-11Cr-2.5Al	2.62		5	75		200	21.053	16.50	12	50.0		21.0		42
P-45	22-5	Ti6Al-4V	5.3	6	12	150		500	54.62	20.75	16	146.25		24.053		16.25
P-46	22-13	Ti-13V-11Cr-4Al	5.65	7	9	45		70	5.75	12.61	12.6	80.31		29.61		37.2
P-47	22-2	Ti6Al-4V	5.75		5	225		600	(Turnings) 32.81	18.86	13	102.0		12.94		12.7
P-48	22-1	Ti6Al-4V	4.05		9	80		275	31.0	14.75	16	61.75		14.56		16.65
P-49	22-4	Ti6Al-4V	1.3		5	70		175	20.8	12.86	None	31.0		16.063		51.75
P-50	22-5	Ti6Al-4V	6.36	6	12	200		550	34.0	21.66	17	121.62		25.62		21
P-51	22-5	Ti6Al-4V	6.2		9	55		250	20.19	25.19	15	69.0		16.25		23.5
P-52	22-13	Ti-13V-11Cr-4Al	2.05		12	60		200	17.0	11.0	None	46.19		21.56		44.75
P-53	22-4	Ti6Al-4V	5.2		9	60		225	37.0	16.56	17	93.86		26.75		28.5
P-54	22-5	Ti6Al-4V	4.3		9	70		800	23.31	17.94	12.5	86.56		14.86		17.2
P-55	22-1	Ti-13V-11Cr-2.5Al	5	7	9	70		175	11.19	13.56	11	83.62		32.66		39.2
P-56	22-2	Ti6Al-4V	6.25	6	9	150		800	23.75	14.81	14.5	98.75		12.15		12.3
P-57	22-2	Ti6Al-4V	5.5	6	9	125		600	27.81	20.0	14	92.81		12.37		13.35

TABLES

TABLE J1 (Continued)

MELTING RECORD

Heat No.	Order No.	Material	MELTING			VACUUM - MICRONS			SKULL WT.		Fluidity Spiral	Gross Heat Mt.		Net Heat Mt.	Percent Humidity Field Percent
			Time	Temp.	Atm.	Start	Stop	Leak Rate	Start	Finish		Heat Mt.	Heat Mt.		
P-5E	22-1	T16A1-WV	12,000	40	1.35	9	150	400	24.19	16.88	None	18.44	11.31	61.5	
P-5E	22-23	CP T1	12,000	40	4.2	6.5	9	50	10.31	19.37	None	62.75	5.56	8.85	
P-6E	22-23	CP T1	12,000	40	4.5	6.5	9	60	11.0	15.69	None	56.94	5.56	9.78	
P-6E	22-11	T130V-11CP	12,000	42	2.35	9	60	150	21.0	10.62	None	49.94	23.31	46.75	
P-6E	22-10	T130V-11CP	12,000	36	5.30	7	12	70	11.0	13.13	10	86.62	32.94	37	
P-6E	22-1	T15A1-WV	12,000	40	5.25	9	70	225	29.31	18.0	6.5	102.13	20.69	20.2	
P-6E	22-15	T17A1-3MC	12,000	40	8.62	7	70	100	2.31	15.25	23	88.62	30.69	34.6	
P-6E	22-16	T15A1-5PV 1.5% Fe, 2 Sn 25 Cu	10,000	40	7.80	7	50	100	9.50	22.75	13	84.94	29.69	35	
P-6E	22-1F	T130V-11CP	12,000	40	5.20	9	55	20	26.56	5.75	9	60.75	26.69	44	
P-6E	22-5	T16A1-WV	12,000	42	6.40	12	250	350	40.0	19.19	None	132.75	15.0	11.3	
P-6E	22-15	T17A1-3MC	2,000	36	4.80	7	60	15	31.37	20.25	18.5	93.62	25.75	31.6	
P-6E	22-16	T15A1-5PV 1.5% Fe, 2 Sn 25 Cu	12,000	40	4.75	7 + Recycle	9	30	32.75	20.0	17	93.37	30.62	32.5	
P-6E	22-1	T16A1-WV	12,000	40	5.6	Recycle	9	150	30.52	16.62	14	105.31	17.063	16.2	
P-6E	22-2	T16A1-WV	12,000	40	5.70	Recycle	9	80	32.37	15.0	16	96.37	17.56	17.85	
P-6E	22-5	T16A1-WV	12,000	40	3.00	Recycle	9	35	17.50	11.86	None	55.44	15.063	27.2	
P-6E	22-5	T16A1-WV	12,000	40	5.45	Recycle	9	100	26.19	17.19	14.5	97.61	16.50	16.9	
P-6E	22-1	T16A1-WV	12,000	40	4.45	7	9	45	26.59	17.50	None	64.56	26.25	31.0	
P-6E	22-1	T16A1-WV	12,000	40	4.63	6	9	50	26.52	17.0	None	63.55	25.31	30.2	
P-6E	22-1	T16A1-WV	12,000	40	4.80	6	9	25	26.81	17.56	None	85.56	21.94	25.6	
P-6E	22-1	T16A1-WV	12,000	40	5.85	7	9	30	27.0	16.86	None	84.86	22.37	26.6	
P-6E	22-1	T16A1-WV	12,000	40	5.12	Recycle	9	30	25.0	14.31	None	87.50	30.75	35.2	
P-6E	22-1	T16A1-WV	12,000	40	5.10	Recycle	9	25	24.0	14.69	None	55.37	27.56	32.3	
P-6E	22-1	T16A1-WV	12,000	40	1.90	Recycle	9	55	30.44	11.61	None	55.0	10.25	25.3	
P-6E	22-1	T16A1-WV	12,000	40	1.85	Recycle	9	60	16.0	12.15	None	29.37	8.75	29.8	
P-6E	22-1	T16A1-WV	12,000	40	5.00	Recycle	9	35	28.0	14.75	None	82.69	25.44	30.6	
P-6E	22-1	T16A1-WV	12,000	40	1.95	Recycle	9	26	14.75	11.13	None	31.62	9.21	30.13	
P-6E	22-1	T16A1-WV	12,000	40	5.55	Recycle	9	50	27.0	13.81	None	85.75	25.19	29.6	
P-6E	22-1	T16A1-WV	12,000	40	1.95	Recycle	9	50	15.0	9.75	None	30.19	7.81	25.8	

TABLES

TABLE J1 (Continued)

MELTING RECORD

Heat No.	Core No.	Mold No.	MELTING		PISTONS			SKULL WT.		Fluidity	Gross		Net	Percent Humidity	
			Time	Elect.	Crucible	Start	During	After	5 Min.		Spinal	Heat Wt.		Yield	Percent
			min.	D.A.	Dia. in.	Start	Weight	Feet	Lead Rate	lb.	in.	lb.	lb.		
1-36	1-1	1-1	1:00	42	5	80	70-160	160	5-15	16.0	None	20.75	5.62	46.4	
1-37	1-1	1-1	1:00	42	5	150	90-330	750	30-37	16.0	None	57.56	14.56	14.5	
1-38	1-1	1-1	1:00	42	5	50	120-220	220	24-0	14.75	None	51.063	24.50	25.6	
1-39	1-1	1-1	1:00	42	5	50	150-220	220	16.0	12.25	None	21.66	5.62	45.7	
1-40	1-1	1-1	1:00	42	5	100	150	400	12.13	15.0	None	75.15	21.063	22.0	
1-41	1-1	1-1	1:00	42	5	45	150	500	16.0	11.15	None	22.0	See foot note (1)		
1-42	1-1	1-1	1:00	42	5	100	200-300	300	24-0	11.15	None	21.063	21.063	21.2	
1-43	1-1	1-1	1:00	42	5	100	100-300	300	14/min.	22.0	None	51.15	19.32	21.6	
1-44	1-1	1-1	1:00	42	5	100	100-300	300	15/min.	17.0	None	21.15	21.15	21.2	
1-45	1-1	1-1	1:00	42	5	45	150	170	17.25	13.15	None	65.66	24.66	25.0	
1-46	1-1	1-1	1:00	42	5	125	150	500	20.0	11.25	15.375	62.0	13.15	15.9	
1-47	1-1	1-1	1:00	42	5	100	100-250	250	14-0	10.15	None	21.0	21.0	20.5	
1-48	1-1	1-1	1:00	42	5	100	100-250	250	26.56	9.31	None	24.56	15.54	16.5	
1-49	1-1	1-1	1:00	42	5	50	175	150	3/min.	16.06	None	4.15	5.50	34.5	
1-50	1-1	1-1	1:00	42	5	65	250	600	14.0	2.25	14	73.54	21.063	31.6	
1-51	1-1	1-1	1:00	42	5	45	150	60	17.50	12.50	None	74.50	24.65	33.2	
1-52	1-1	1-1	1:00	42	5	100	150	220	7/min.	14.81	15.5	75.61	13.063	16.4	
1-53	1-1	1-1	1:00	42	5	150	200	300	10/min.	13.56	None	75.61	21.25	22.0	
1-54	1-1	1-1	1:00	42	5	150	245	400	10/min.	16.0	16.75	75.15	10.31	13.0	
1-55	1-1	1-1	1:00	42	5	75	180	240	41/min.	22.50	None	75.50	24.66	33.0	
1-56	1-1	1-1	1:00	42	5	80	190	500	22/min.	19.25	17.0	75.063	22.81	29.0	
1-57	1-1	1-1	1:00	42	5	125	240	400	21/min.	32.25	22.50	133.56	23.063	17.22	
1-58	1-1	1-1	1:00	42	5	60	225	300	35/min.	22.25	17.75	10.50	79.66	16.56	
1-59	1-1	1-1	1:00	42	5	75	180	400	27/min.	23.75	15.25	61.37	20.94	25.75	
1-60	1-1	1-1	1:00	42	5	135	165	450	5/min.	21.37	13.063	See note (2)	See note (2)		
1-61	1-1	1-1	1:00	42	5	140	250	450	16/min.	24.44	16.0	64.37	25.94	24.6	
1-62	1-1	1-1	1:00	42	5	65	165	180	7/min.	20.0	12.61	32.62	4.62	10.31	
1-63	1-1	1-1	1:00	42	5	85	245	450	5/min.	18.063	None	77.37	21.37	27.60	
1-64	1-1	1-1	1:00	42	5	55	200	400	17/min.	16.0	13.0	53.0	22.37	24.0	
1-65	1-1	1-1	1:00	42	5	40	250	300	14/min.	17.0	115.44	86.0	6.62	9.8	
1-66	1-1	1-1	1:00	42	5	65	175	250	5/min.	15.0	11.56	52.19	11.0	21.1-Reject	
1-67	1-1	1-1	1:00	42	5	25	175	250	19/min.	15.0	16.50	54.50	24.65	26.1	
1-68	1-1	1-1	1:00	42	5	60	250	400	6/min.	32.25	14.0	22.56	22.37	26.1	
1-69	1-1	1-1	1:00	42	5	100	175	345	12/min.	20.19	18.0	64.50	9.62	28.30	
1-70	1-1	1-1	1:00	42	5	45	125-150	200	27/min.	22.56	15.0	87.13	24.65		

Foot Note: 1. Shell Core collapsed when poured.

2. Contaminated scrap. Plug in mold came loose.

TABLES

TABLE J1 (Continued)

MELTING RECORD

Heat No.	Crater No.	Material	MELTING			VAJUNK - MICRONS			SKULL WT.			Fluidity Spiral	Gross Heat Wt. Lb.	Net Heat Wt. Lb.	Percent Yield	Humidity Percent	
			Time Min.	Elect. Dia.	Crucible	Start	After	5 Min	Start	Fin	Sh						
			Avgs	Volts		Temp	Rate	Leak	Rate	Rate	Rate						
P-131	22-1	T-161 (new)	12,000	40	6.15	Recycle	100	175	350	24/min.	16.62	15.86	None	92.15	22.37	24.2	
P-132	22-2	T-161 (new)	12,000	40	5.80	Recycle	100	175	400	26/min.	18.62	12.50	None	84.0	23.19	27.5	
P-133	22-3	T-161 (new)	12,000	40	6.75	Recycle	60	130	410	27/min.	16.0	16.19	None	85.0	22.69	25.6	
P-134	22-4	T-161 (new)	12,000	40	5.75	Recycle	70	160	350	30/min.	19.0	16.37	None	86.50	22.063	25.6	
P-135	22-5	T-161 (new)	12,000	40	5.30	Recycle	70	160	350	12/min.	25.0	13.0	None	84.25	21.75	25.6	
P-136	22-6	T-161 (new)	12,000	40	6.15	Recycle	175	175-240	400	12/min.	17.0	11.25	None	87.50	21.81	25.6	
P-137	22-7	T-161 (new)	12,000	40	5.20	Recycle	60	160	250	24/min.	30.19	11.37	None	75.37	23.62	23.6	
P-138	22-8	T-161 (new)	12,000	40	6.30	Recycle	60	175	500	23/min.	19.66	14.19	17.750	99.75	24.25	24.2	
P-139	22-9	T-161 (new)	12,000	40	6.30	Recycle	60	175	500	2/min.	35.0	17.50	16.50	99.75	25.37	25.6	
P-140	22-10	T-161 (new)	12,000	40	6.40	Recycle	45	175	350	21/min.	15.0	25.0	None	79.0	11.13	14.2	
P-141	22-11	T-161 (new)	12,000	40	5.64	Recycle	250	350	22/min.	25.0	13.69	None	92.62	21.69	23.40		
P-142	22-12	T-161 (new)	12,000	40	6.52	Recycle	40	160	600	14/min.	23.0	13.69	None	86.37	25.13	25.2	
P-143	22-13	T-161 (new)	12,000	40	5.46	Recycle	175	250									
No recycle.																	
P-144	22-14	Reject Electrode broke															
P-145	22-15	T-161 (new)	12,000	40	5.90	Recycle	5	55	350	5/min.	12.81	22.56	16.0	115.37	12.94	11.1	
P-146	22-16	T-161 (new)	12,000	40	6.20	Sample Slides C	5	Electrode	Broke	did not pour							
P-147	22-17	T-161 (new)	12,000	40	5.66	7	5	75	120	155	10.13	14.44	None	61.13	23.62	35.6	
P-148	22-18	T-161 (new)	12,000	40	6.23	7	5	60	125	16/min.	10.0	16.0	4.5	80.0	23.44	25.2	
P-149	22-19	T-161 (new)	12,000	40	3.44	Recycle	5	55	200	235	17.37	10.13	None	39.55	5.13	12.9	
P-150	22-20	T-161 (new)	12,000	40	5.51	Recycle	4	55	200	600	40.0	33.0	15.5	100.50	17.44	16.6	
P-151	22-21	T-161 (new)	12,000	40	5.72	Recycle	4	250	215	750	15/min.	15.19	None	82.81	23.37	37.5	
P-152	22-22	T-161 (new)	12,000	40	5.85	Recycle	12	160	200	300	30/min.	36.37	16.44	None	11.13	23.56	4.7
P-153	22-23	T-161 (new)	12,000	40	5.45	7	5	150	200	250	10.25	5.0	None	46.25	21.13	44.7	
P-154	22-24	T-161 (new)	12,000	40	1.45	Recycle	5	120	175	200	14.0	6.94	None	16.44	9.37	50.1	
P-155	22-25	T-161 (new)	12,000	40	3.80	7	5	260	190	250	9.50	6.25	16	36.94	11.62	30.5	
P-156	22-26	T-161 (new)	12,000	40	3.72	7	5	160	50	150	10/min.	10.0	6.64	22	50.0	22.44	
P-157	22-27	T-161 (new)	12,000	40	3.97	7	5	150	80	230	3/min.	10.0	8.94	24	56.13	20.6	
P-158	22-28	T-161 (new)	12,000	40	6.10	7	5	90	140	210	7/min.	10.0	14.56	17.25	82.0	22.063	26.9
P-159	22-29	T-161 (new)	12,000	40	4.26	7	5	100	185	250	9.0	8.13	14.75	62.37	11.25	17.56	
P-160	22-30	T-161 (new)	12,000	40	4.90	7	5	90	150	250	6.0	10.56	22	61.69	13.69	22.1	
P-161	22-31	T-161 (new)	12,000	40	4.75	5 x 14	5	200	250	750	6.50	9.37	15	62.0	14.69	23.6	
P-162	22-32	T-161 (new)	12,000	40	5.40	Recycle	12	350	400	2/min.	22.50	16.23	None	86.86	25.37	26.6	
P-163	22-33	T-161 (new)	12,000	40	5.40	Recycle	12	90	200	300	6/min.	40.0	22.81	None	96.37	25.44	26.4
P-164	22-34	T-161 (new)	12,000	40	5.20	Recycle	5	200	400	1,000	22.37	15.81	None	92.0	11.063	12.1	
P-165	22-35	T-161 (new)	12,000	40	5.10	Recycle	5	200	265	400	5/min.	27.05	14.56	13	90.44	10.62	11.6
P-166	22-36	T-161 (new)	12,000	40	6.36	Recycle	12	250	250	425	3/min.	28.0	18.15	None	100.50	13.62	12.9
P-167	22-37	T-161 (new)	12,000	40	5.20	Recycle	12	130	250	300	3/min.	33.31	19.063	None	98.44	13.44	13.7
P-168	22-38	T-161 (new)	12,000	40	5.55	Recycle	12	300	400	1,000	28.9	22.67	None	105.13	12.56	11.9	
P-169	22-39	T-161 (new)	12,000	40	3.06	Recycle	5	15	100	200	2/min.	11.56	19.65	None	34.44	5.62	16.0

TABLES

TABLE J1 (Continued)

MELTING RECORD

Heat No.	Jroor. No.	Material	MELTING			VACUUM - MICRONS			SKULL WT.		Fluidity Spiral	Gross Heat Wt.		Percent Humidity
			Time Min.	Elect. Dia.	Crucible Dia. in.	Start	Melt	After Four	5 min. Leak Rate	Start Finish Lb.		Lb.	Net Heat Wt. Lb.	
P-104	22-4	Ti-6Al-4V	12,000	42	3.05	Recycle	9	35	100	200	11.56	19.69	34.44	None
P-105	22-1	Ti-6Al-4V	12,000	42	5.69	Recycle	12	275	400	750	16.50	14.75	51.88	None
P-106	22-1	Ti-6Al-4V	12,000	42	6.32	Recycle	9	175	225	300	16.75	13.75	76.37	None
P-107	22-1	Ti-6Al-4V	12,000	42	6.25	Recycling	9	25	250	500	24.62	19.31	96.13	None
P-108	22-35	50%-60%-2%	12,500	40	2.57	7	40	50	90	9.0	9.0	9.0	40.31	None
P-109	22-25	20%-20%-2%	12,500	40	4.35	7	70	60	80	3/min.	10.0	13.25	59.44	None
P-110	22-22	60%-10%-1%	12,500	42	4.45	7	30	40	40	5/min.	10.0	9.69	63.44	None
P-111	22-1	Ti-6Al-4V	12,000	40	6.40	Rectang.	9	50	125	300	19.75	16.85	87.69	None
P-112	22-1	Ti-6Al-4V	12,000	42	5.20	Rectang.	9	200	250	500	25.62	17.69	91.37	None
P-113	22-1	Ti-6Al-4V	12,000	42	5.60	Bar Str	9	35	250	600	22.50	10.75	97.0	None
P-114	22-1	Ti-6Al-4V	10,000	40	2.25	Recycle	9	70	-	12/min.	14.62	56.50	Elect.	None
P-115	22-1	Ti-6Al-4V	12,500	42	6.37	Recycle	5	60	200	750	14.50	11.56	64.50	None
P-116	22-1	Ti-6Al-4V	12,500	42	6.56	Recycle	9	250	350	850	14.61	13.31	87.68	None
P-117	22-1	Ti-6Al-4V	12,500	42	6.34	Recycle	9	190	250	750	13.19	13.75	81.0	None
P-118	22-1	Ti-6Al-4V	12,000	42	5.31	Recycle	9	165	500	1,000	35.0	18.50	95.75	None
P-119	22-1	Ti-6Al-4V	12,000	44	3.98	Recycle	9	200	350	1,000	44.56	14.063	90.44	None
P-120	22-1	Ti-6Al-4V	11,000	42	4.95	Recycle	9	150	350	1,000	26.19	14.13	83.25	None
P-121	22-1	Ti-6Al-4V	12,000	42	4.35	Recycle	12	100	250	350	23.81	22.50	66.063	None
P-122	22-1	Ti-6Al-4V	12,000	42	4.39	Recycle	9	100	150	200	13.69	20.69	70.50	None
P-123	22-1	Ti-6Al-4V	12,000	42	5.21	Recycle	12	25	110	250	28.50	25.19	87.44	None
P-124	22-2c	100% Ti	12,000	36	5.00	7	45	95	150	4/min.	10.50	8.81	74.31	None
P-125	22-31	12 Mo Ti	11,000	42	2.73	7	60	50	100	6/min.	12.19	7.75	39.44	6
P-126	22-4c	3 Mo Ti Se	12,000	42	3.61	7	45	50	100	3/min.	11.62	22.56	40.19	None
P-127	22-4c	6Al-4V-15i	12,500	40	4.13	7	70	60	200	5/min.	11.56	9.94	55.37	10
P-128	22-1	6Al-4V	12,000	42	6.32	Recycle	9	225	300	1,000	27.62	14.50	99.0	None
P-129	22-1	6Al-4V	11,250	42	6.50	Recycle	9	105	300	1,000	17.25	13.50	90.75	None
P-130	22-1	6Al-4V	12,500	42	6.80	Recycle	9	160	600	750	17.25	20.13	99.19	None
P-131	22-5	6Al-4V	12,000	42	3.32	Recycle	9	125	250	750	25.0	10.44	47.063	None
P-132	22-1	6Al-4V	13,000	42	2.62	Recycle	9	70	250	1,000	17.0	8.19	31.94	None
P-133	22-3c	6Al-4V	13,000	42	4.0	7	50	25	95	7/min.	11.0	14.44	54.25	14
P-134	22-3	6Al-4V	13,000	40	3.37	7	90	70	150	20/min.	13.0	13.37	65.81	1.7
P-135	22-1	6Al-4V	13,500	42	10.07	Recycle	12	250	400	1,000	30.0	37.56	193.94	None
P-136	22-3	6Al-4V	12,000	36	3.31	7	50	35	85	6/min.	10.0	8.86	39.063	9
P-137	22-3c	20%-20%	12,500	40	2.92	7	60	70	90	15/min.	12.0	6.81	41.69	11
P-138	22-1	6Al-4V	14,000	42	10.86	8	12	180	400	5/min.	44.13	45.56	92.0	15
P-139	22-1	6Al-4V	12,500	41	4.66	Recycle	12	200	450	1,000	46.13	33.13	93.175	None
P-140	22-1	6Al-4V	12,000	42	5.25	Recycle	12	200	450	1,000	45.37	33.56	102.0	None
P-141	22-1	6Al-4V	14,000	42	4.99	Recycle	9	100	350	700	30.0	15.0	94.37	17
P-142	22-6	6Al-4V	13,000	42	5.14	Recycle	9	45	400	1,000	25.0	13.31	97.56	15
P-143	22-6	6Al-4V	13,000	42	5.44	Recycle	9	150	300	500	30.0	19.0	85.44	14
P-144	22-1	6Al-4V	12,500	42	4.79	Recycle	12	175	300	750	34.50	27.50	94.50	None

TABLES

TABLE J1 (Continued)

MELTING RECORD

Heat No.	Order No.	Material	MELTING			VACUUM - MICRONS			SKULL WT.		Fluidity Spiral In.	Gross Heat Wt. Lb.		Net Heat Wt. Lb.		Percent Humidity Yield Percent	
			Time Min.	Elect. Dia.	Crucible Dia. In.	Start	During Melt	After Pour	Lead Rate 5 Min.	Start Finish Lb.							
P-22-1	22-1	6A1-4W	8.34	6	12	175	350	1,000+	6/min.	34.62	14.69	81.19	23.0	12.7	27		
P-22-2	22-2	T1-20A	2.61	7	9	30	60	70	6/min.	9.54	9.56	37.75	6.50	17.2	44		
P-22-3	22-3	T1-15A-2B	2.34	7	9	40	65	100	5/min.	10.0	7.37	40.54	9.0	22.0	39		
P-22-4	22-4	T1-25A	2.66	7	9	60	30	100	7/min.	11.0	5.69	40.13	8.063	22.6	37		
P-22-5	22-5	T1-25A	2.75	7	9	45	50	70	7/min.	11.56	7.50	41.13	5.13	21.9	32		
P-22-6	22-6	6A1-4W	10.17	Recycle	12	160	420	1,000	5/min.	44.25	32.88	56.50	23.50	12.1	28		
P-22-7	22-7	6A1-4W	2.65	Recycle	9	100	300	400	20/min.	13.31	7.31	30.56	9.50	46.4	36		
P-22-8	22-8	6A1-4W	4.66	Recycle	12	60	250	400	20/min.	24.81	15.0	90.25	26.88	39.8	28		
P-22-9	22-9	6A1-4W	9.55	Recycle	12	50	300	725	5/min.	34.50	30.81	176.44	23.25	13.0	29		
P-22-10	22-10	6A1-4W	6.15	Recycle	9	50	300	725	11/min.	25.0	15.69	55.37	11.56	12.1	31		
P-22-11	22-11	6A1-4W	5.36	Recycle	9	90	275	500	4/min.	15.31	14.25	70.31	15.0	21.3	30		
P-22-12	22-12	6A1-4W	5.72	Recycle	9	60	275	400	6/min.	13.25	12.44	66.37	27.37	30.9	36		
P-22-13	22-13	6A1-4W	4.47	Recycle	9	100	350	550	6/min.	15.0	10.0	55.37	8.063	12.3	36		
P-22-14	22-14	6A1-4W	4.70	Recycle	9	32	250	500	1 1/2 in.	18.13	12.81	88.81	25.94	25.2	33		
P-22-15	22-15	6A1-4W	4.64	Recycle	9	30	300	310	8/min.	22.50	14.13	68.86	27.25	20.7	42		
P-22-16	22-16	6A1-4W	2.2	Recycle	9	35	200	250	9/min.	12.0	5.88	19.37	5.44	28.3	36		
P-22-17	22-17	6A1-4W	6.16	Recycle	9	125	350	1,000	6/min.	19.13	5.44	52.81	10.50	11.3	37		
P-22-18	22-18	6A1-4W	5.00	Recycle	9	100	300	1,000	10/min.	24.37	13.19	51.85	10.62	11.6	34		
P-22-19	22-19	6A1-4W	10.	Recycle	12	110	255	720	7/min.	17.88	18.75	148.75	17.88	11.6	34		
P-22-20	22-20	6A1-4W	6.04	7	12	325	500	1,000	17/min.	24.0	17.69	130.96	21.25	15.6	34		
P-22-21	22-21	6A1-4W	3.35	7	9	30	60	125	7/min.	17.0	3.25	34.75	11.13	20.3	44		
P-22-22	22-22	6A1-4W	2.75	7	9	75	100	150	6/min.	11.0	5.25	37.75	5.063	13.4	45		
P-22-23	22-23	6A1-4W	5.45	Recycle	9	50	350	700	4/min.	33.0	20.81	62.65	9.44	11.4	30		
P-22-24	22-24	6A1-4W	6.52	Recycle	9	150	275	450	11/min.	23.0	19.62	95.94	13.063	14.2	40		
P-22-25	22-25	6A1-4W	3.7	7	9	120	175	350	10/min.	10.61	9.50	46.44	8.13	17.5	35		
P-22-26	22-26	6A1-4W	4.8	Recycle	9	55	260	500	7/min.	17.88	12.66	64.13	15.50	24.2	35		
P-22-27	22-27	6A1-4W	2.45	Recycle	9	30	175	1,000	8/min.	20.15	63.56	Electrode Broke - No pour	5.50	13.3	36		
P-22-28	22-28	6A1-4W	4.5	7	9	105	275	500	4/min.	17.0	11.063	71.31	5.50	13.3	36		
P-22-29	22-29	115m-102m-2A	4.5	7	9	125	160	210	7/min.	10.0	9.25	54.44	9.62	16.7	37		
P-22-30	22-30	6A1-4W	2.56	Recycle	9	45	200	1,000	9/min.	20.0	11.69	38.75	13.50	34.6	36		
P-22-31	22-31	6A1-4W	4.65	Recycle	9	60	250	600	15/min.	17.69	11.75	72.88	10.0	13.7	35		
P-22-32	22-32	5Mm-6Mm-2Hm-2A	2.50	7	9	40	50	100	7/min.	9.0	10.0	38.0	8.50	22.4	26		
P-22-33	22-33	6A1-4W	3.44	7	9	50	50	500	7/min.	6.0	5.56	55.62	9.50	17.0	34		
P-22-34	22-34	6A1-4W	4.95	Recycle	9	65	200	500	11/min.	19.50	11.81	68.50	9.13	13.3	36		
P-22-35	22-35	115m-102m-4A	3.56	7	9	50	100	250	5/min.	12.0	7.62	55.063	8.62	15.6	32		
P-22-36	22-36	6A1-4W	3.25	Recycle	9	90	180	500	6/min.	20.0	70.69	Electrode Broke - No pour	15.063	20.4	32		
P-22-37	22-37	6A1-4W	4.90	Recycle	9	80	275	530	6/min.	28.0	10.25	73.62	15.063	20.4	32		
P-22-38	22-38	6A1-4W	6.56	Recycle	9	35	250	1,000	10/min.	20.0	18.063	100.81	19.13	19.0	35		
P-22-39	22-39	6A1-4W	4.49	Recycle	9	85	300	750	6/min.	25.0	14.37	87.75	12.13	13.8	33		

TABLES

TABLE J1 (Continued)

MELTING RECORD

No.	Date	Mater.	MELTING		Time	Elect. Dia.	Crucible Dia.	VACUUM - MICRONS		SKULL WT.		Fluidity Spiral	Gross Heat Wt. Lb.	Net Heat Wt. Lb.	Percent Purity			
			Volts	Amperes				Start	Stop	Start	Finish							
1-1	1-1	DAI-100	43	14,000	6.75	Recycle	9	105	375	1,000	5/min.	11.13	14.37	15	100.0	11.19	10.9	34
1-2	1-1	DAI-100	43	13,000	5.95	Recycle	9	75	325	1,000	4/min.	17.25	20.44	13	100.31	11.37	11.3	40
1-3	1-1	DAI-100	43	15,000	6.14	7	12	75	300	1,000	13/min.	17.69	14.37	20	10.56	11.61	10.3	33
1-4	1-1	DAI-100	43	14,000	5.60	Recycle	14	425	500	1,000	25/min.	27.37	19.86	18	102.50	10.69	10.4	36
1-5	1-1	DAI-100	43	11,000	7.06	7	12	30	37	69	6/min.	11.50	19.37	None	134.063	117.37	67.3	34
1-6	1-1	DAI-100	43	14,000	4.9	Recycle	12	50	300	750	2/min.	22.31	12.50	15	93.19	8.86	9.5	42
1-7	1-1	DAI-100	43	11,000	6.20	Recycle	9	80	300	750	5/min.	18.75	6.37	14	95.44	9.0	9.0	36
1-8	1-1	DAI-100	43	14,000	5.25	Recycle	9	80	250	700	6/min.	23.0	6.13	14	87.44	4.50	10.3	40
1-9	1-1	DAI-100	43	14,000	3.1	3	5	37	5	3	6/min.	6.0	6.31	None	46.063	4.50	9.7	29
1-10	1-1	DAI-100	43	13,000	5.10	Recycle	14	45	250	1,000	2.4/min.	17.0	7.0	None	102.86	19.3	16.6	43
1-11	1-1	DAI-100	43	14,000	3.90	Recycle	9	35	250	500	6/min.	10.0	6.56	13	56.61	9.15	16.1	35
1-12	1-1	DAI-100	43	14,000	3.24	Recycle	9	40	75	100	1.6/min.	9.25	6.053	None	36.62	4.44	11.6	15
1-13	1-1	DAI-100	43	11,000	5.67	Recycle	9	75	300	1,000	4/min.	20.0	15.25	None	68.75	7.37	6.2	34
1-14	1-1	DAI-100	43	12,000	3.10	Recycle	9	80	200	300	7/min.	13.13	8.37	None	30.0	2.50	6.3	34
1-15	1-1	DAI-100	43	12,000	3.10	Recycle	9	25	8	115	6/min.	16.37	5.61	None	31.19	4.59	13.0	34
1-16	1-1	DAI-100	43	14,000	4.51	Recycle	9	100	250	750	5/min.	22.0	16.75	None	73.50	36.69	49.5	4
1-17	1-1	DAI-100	43	12,000	5.20	Recycle	9	68	275	750	3/min.	30.0	13.053	None	44.15	1.69	1.9	22
1-18	1-1	DAI-100	43	12,000	2.70	Recycle	9	50	200	200	10/min.	7.50	3.053	None	17.50	3.66	14.6	34
1-19	1-1	DAI-100	43	12,000	1.6	Recycle	5	35	51	112	1/min.	4.25	4.37	None	27.13	3.66	20.4	34
1-20	1-1	DAI-100	43	13,000	6.49	Recycle	12	150	300	1,000	5/min.	21.62	16.50	10	56.31	22.15	20.4	4
1-21	1-1	DAI-100	43	12,500	3.90	Recycle	9	125	250	1,000	5/min.	16.0	9.15	None	46.13	12.75	21.6	33
1-22	1-1	DAI-100	43	11,000	6.07	Recycle	12	70	300	900	7/min.	21.50	17.44	None	67.37	13.44	17.7	33
1-23	1-1	DAI-100	43	12,000	4.7	Recycle	9	60	250	500	4/min.	20.0	10.62	None	67.19	5.44	14.0	34
1-24	1-1	DAI-100	43	12,000	4.50	Recycle	9	75	250	400	3/min.	24.50	13.50	15	70.69	12.13	17.1	32
1-25	1-1	DAI-100	43	12,000	5.33	Recycle	9	35	250	600	7/min.	23.0	15.13	5	65.69	6.54	10.4	30
1-26	1-1	DAI-100	43	14,000	1.90	Recycle	9	110	200	275	7/min.	7.50	4.75	None	12.44	2.44	19.5	31
1-27	1-1	DAI-100	43	14,000	2.6	Recycle	9	51	56	92	4/min.	9.0	6.0	None	35.063	4.50	12.6	31
1-28	1-1	DAI-100	43	12,500	1.6	Recycle	9	125	175	275	9/min.	7.0	4.56	None	13.31	2.50	20.3	34
1-29	1-1	DAI-100	43	12,500	5.5	Recycle	9	60	300	450	7/min.	21.0	15.94	None	63.50	25.44	26.5	40
1-30	1-1	DAI-100	43	14,500	1.3	Recycle	9	50	175	200	5/min.	7.0	4.61	None	11.62	2.44	20.5	30
1-31	1-1	DAI-100	43	14,500	2.45	Recycle	9	200	150	175	4/min.	8.37	6.51	None	32.75	7.37	13.3	34
1-32	1-1	DAI-100	43	14,500	1.95	Recycle	9	40	200	250	7/min.	8.0	4.86	None	13.0	2.50	19.2	34
1-33	1-1	DAI-100	43	14,500	5.75	Recycle	9	80	240	400	5/min.	20.0	17.13	None	65.69	25.44	29.7	39
1-34	1-1	DAI-100	43	12,000	6.30	Recycle	9	45	200	300	3/min.	20.0	7.053	None	33.50	33.50	24.6	34
1-35	1-1	DAI-100	43	13,000	6.30	Recycle	9	120	175	1,000	5/min.	1.50	2.62	None	93.69	13.94	14.6	42
1-36	1-1	DAI-100	43	14,500	1.55	Recycle	9	70	125	175	4/min.	7.0	4.68	None	10.69	1.15	21.1	42
1-37	1-1	DAI-100	43	14,500	3.40	Recycle	9	75	280	300	5/min.	17.0	10.94	None	42.82	3.11	19.7	33
1-38	1-1	DAI-100	43	14,500	6.40	Recycle	9	200	350	400	15/min.	26.0	16.0	None	92.30	1.11	14.3	42
1-39	1-1	DAI-100	43	14,500	6.30	Recycle	9	200	350	400	15/min.	26.0	16.0	None	92.30	1.11	14.3	42
1-40	1-1	DAI-100	43	14,500	1.55	Recycle	9	125	175	1,000	5/min.	13.75	5.69	None	26.13	4.50	17.2	39

TABLES

TABLE J1 (Continued)

MELTING RECORD

Time	Temp	Wt	MELTING		VACUUM - MICRONS	5 Min. Leak Rate	SKULL WT.		Fluidity	Gross Weight	Net Weight	Percent Humidity
			Time	Elect. Dig. in.	Start	During	Start	Finish				
			Min.			Min.				lb.	lb.	Percent
1	12.00	4	5.50	Recycle	75	300	700	3/min.	21.0	15.44	65.75	19.6
2	12.00	4	1.42	Recycle	100	150	200	5/min.	11.0	5.0	25.0	24.3
3	12.00	4	1.30	Recycle	150	200	250	4/min.	7.0	4.62	13.0	24.3
4	12.00	4	3.25	Recycle	125	200	1,000	7/min.	25.0	6.37	42.69	30.1
5	12.00	4	5.70	Recycle	75	300	500	7/min.	6.0	8.13	96.25	10.0
6	12.00	4	5.16	5 sec.	125	450	750	7/min.	8.0	8.13	96.25	5.7
7	12.00	4	6.35	Recycle	45	200	750	5/min.	17.0	16.66	52.0	12.3
8	12.00	4	5.40	Recycle	65	450	400	5/min.	16.75	14.75	78.84	16.6
9	12.00	4	5.40	Recycle	80	300	750	3/min.	25.0	16.96	82.82	16.6
10	12.00	4	5.40	Recycle	40	120	145	4/min.	5.66	4.75	11.82	22.4
11	12.00	4	5.40	Recycle	100	375	1,000	10/min.	5.0	23.3	112.15	17.0
12	12.00	4	7.11	Recycle	50	250	1,000	4/min.	25.0	24.62	130	7.3
13	12.00	4	5.40	Recycle	50	250	800	4/min.	25.0	24.62	90.1	25.3
14	12.00	4	5.40	Recycle	125	250	750	5/min.	25.0	13.0	105.3	24.3
15	12.00	4	5.40	Recycle	40	200	600	5/min.	25.0	13.31	52.0	42.6
16	12.00	4	5.40	Recycle	50	250	550	5/min.	25.0	5.15	55.0	16.0
17	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
18	12.00	4	5.40	Recycle	35	25	120	4/min.	10.0	6.13	30.75	32.1
19	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
20	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
21	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
22	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
23	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
24	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
25	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
26	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
27	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
28	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
29	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
30	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
31	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
32	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
33	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
34	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
35	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
36	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
37	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
38	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
39	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
40	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
41	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
42	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
43	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
44	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
45	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
46	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
47	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
48	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
49	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15
50	12.00	4	5.40	Recycle	50	275	375	3/min.	25.0	15.43	56.0	27.15

TABLES

TABLE J1 (Continued)

MELTING RECORD

Head No.	Room No.	Date	MELTING			VACUUM - MICRONS			SKILL WT.		Fluidity Spiral in.	Gross Heat Mt. Lb.		Net Heat Mt. Lb.		Percent Humidity Yield Percent	
			Time Min.	Temp. Volt	Elect. Crucible Dis. in.	Start	During Melting	5 Min. Leak Rate	Start Lb.	Finish Lb.							
P121	22-1	6-1-44	6:40	44	Recycle	12	50	575	25.0	16.5	None	104.1	19.30	16.5	29		
P122	22-2	6-1-44	6:52	43	Recycle	12	160	500	39.0	26.2	17	123.1	16.80	13.6	32		
P123	22-3	6-1-44	6:72	43	Recycle	12	100	350	22.0	20.1	None	125.1	30.94	25.0	35		
P124	22-4	6-1-44	6:56	45	Recycle	12	115	500	50.0	21.15	None	136.16	28.77	21.1	35		
P125	22-5	6-1-44	7:02	42	Recycle	12	110	450	30.6	22.5	None	127.2	29.98	23.6	31		
P126	22-6	6-1-44	5:40	42	Recycle	12	50	500	23.0	25.4	None	79.7	19.42	15.6	29		
P127	22-7	6-1-44	6:55	42	Recycle	12	50	500	31.0	23.0	None	121.6	26.64	23.5	29		
P128	22-8	6-1-44	6:70	44	Recycle	12	50	500	25.0	18.6	None	116.5	26.27	22.0	37		
P129	22-9	6-1-44	6:80	44	Recycle	12	50	500	17.0	18.19	None	92.50	28.62	31.2	26		
P130	22-10	6-1-44	5:35	43	Recycle	12	110	275	18.50	22.50	15	104.4	9.18	6.75	42		
P131	22-11	6-1-44	3:75	42	Recycle	5	50	175	13.6	6.6	15	42.0	4.63	11.0	36		
P132	22-12	6-1-44	4:75	42	Recycle	5	45	55	6.19	6.44	None	36.10	5.3	14.6	36		
P133	22-13	6-1-44	5:75	43	Recycle	5	90	500	14.0	15.11	None	48.90	16.6	29.1	39		
P134	22-14	6-1-44	1:52	42	Recycle	5	80	175	10.0	6.75	None	25.0	4.50	15.2	30		
P135	22-15	6-1-44	5:50	42	Recycle	5	50	125	20.0	20.36	None	96.14	20.7	23.0	32		
P136	22-16	6-1-44	2:55	42	Recycle	5	50	125	20.0	9.0	None	31.0	11.04	36.8	34		
P137	22-17	6-1-44	5:75	42	Recycle	5	70	275	30.0	14.50	11	56.0	25.86	27.0	40		
P138	22-18	6-1-44	1:50	42	Recycle	5	75	210	25.0	5.87	6	50.94	9.61	18.9	44		
P139	22-19	6-1-44	7:10	42	Recycle	12	125	200	25.0	23.0	31	126.75	11.61	10.7	35		
P140	22-20	6-1-44	6:55	42	Recycle	12	80	400	42.14	35.0	None	132.1	30.78	23.2	30		
P141	22-21	6-1-44	4:55	43	Recycle	12	150	300	35.0	17.50	15	100.24	26.67	27.6	37		
P142	22-22	6-1-44	5:35	42	Recycle	5	45	125	11.0	24.75	18	71.56	12.54	17.5	43		
P143	22-23	6-1-44	4:75	43	Recycle	12	75	175	24.0	16.75	None	86.82	20.78	31.2	29		
P144	22-24	6-1-44	5:42	43	Recycle	12	120	250	20.0	15.37	17	101.1	15.36	15.1	35		
P145	22-25	6-1-44	5:55	43	Recycle	12	90	250	22.0	23.25	18	101.18	10.36	9.76	41		
P146	22-26	6-1-44	5:40	42	Recycle	12	55	150	20.0	18.66	None	90.6	20.65	22.8	32		
P147	22-27	6-1-44	5:05	42	Recycle	5	50	220	6.75	9.31	21	50.65	5.74	19.2	38		
P148	22-28	6-1-44	3:35	42	Recycle	5	45	110	11.0	8.0	12	53.19	9.67	18.2	44		
P149	22-29	6-1-44	3:24	42	Recycle	5	50	155	10.61	8.06	11	51.66	10.04	20.2	44		
P150	22-30	6-1-44	3:5	42	Recycle	5	80	125	12.0	9.50	8	51.75	10.45	20.2	40		
P151	22-31	6-1-44	3:17	42	Recycle	5	65	150	10.0	6.75	22	53.13	9.99	18.6	39		
P152	22-32	6-1-44	4:34	42	Recycle	5	115	130	3.62	6.25	4	55.19	9.55	17.3	38		
P153	22-33	6-1-44	3:00	36	Recycle	5	80	110	11.0	9.93	8	53.0	5.74	18.3	40		
P154	22-34	6-1-44	3:80	36	Recycle	5	65	145	12.94	9.75	16	53.94	9.81	18.1	44		

TABLES

TABLE J1 (Continued)

MELTING RECORD

Heat No.	Crook No.	Material	MELTING			VACUUM - MICRONS			SKULL WT.		Fluidity Spire in.	Gross Heat Wt.		Net Heat Wt.		Percent Humidity	
			Time Min.	Temp. F.	Crucible Dia.	Start	During Melt	After Pour	5 Min. Leak Rate	Start Lb.	Finish Lb.	Heat Wt. Lb.	Yield	Heat Wt. Lb.	Yield	Percent	Humidity
P-20	21-21	SA-1000	4:51	42	Recycle	12	55	310	6/min.	22.37	16.44	81.25	21.72	21.72	25.4	35	35
P-21	21-22	SA-1000	4:50	42	Recycle	12	55	245	1.4/min.	26.25	19.12	88.12	21.67	21.67	24.6	35	35
P-22	21-23	SA-1000	5:55	42	Recycle	12	55	275	5/min.	32.0	18.06	105.65	11.59	11.59	11.0	35	35
P-23	21-24	SA-1000	5:13	62	Recycle	12	42	455	7/min.	25.0	15.56	94.32	6.43	6.43	9.14	3	3
P-24	21-25	SA-1000	5:05	42	Recycle	12	42	310	4/min.	25.0	15.75	97.25	18.97	18.97	15.5	34	34
P-25	21-26	SA-1000	5:50	42	Recycle	12	27	315	4/min.	18.0	16.15	95.31	15.80	15.80	16.6	32	32
P-26	21-27	SA-1000	5:55	42	Recycle	12	45	255	3.8/min.	24.55	16.88	96.65	19.05	19.05	15.8	34	34
P-27	21-28	SA-1000	4:52	42	Recycle	12	100	725	2.1/min.	21.15	16.0	81.82	9.43	9.43	11.6	32	32
P-28	21-29	SA-1000	5:42	42	Recycle	12	33	320	4/min.	26.75	17.62	101.0	10.1	10.1	11.6	32	32
P-29	21-30	SA-1000	6:32	42	Recycle	12	110	290	10/min.	23.0	20.15	106.15	5.0	5.0	4.56	37	37
P-30	21-31	SA-1000	5:35	42	Recycle	5	51	115	2/min.	27.0	17.87	88.35	19.20	19.20	21.7	52	52
P-31	21-32	SA-1000	3:52	42	Recycle	5	100	330	8/min.	21.0	12.85	64.16	21.24	21.24	33.1	33	33
P-32	21-33	SA-1000	5:04	42	Recycle	5	75	335	6/min.	15.50	12.82	70.13	23.40	23.40	33.4	27	27
P-33	21-34	SA-1000	5:04	42	Recycle	5	80	250	6/min.	15.81	10.18	4.55	10.7	10.7	10.7	26	26
P-34	21-35	SA-1000	4:55	42	Recycle	12	70	250	5/min.	25.0	15.56	65.5	16.75	16.75	22.0	26	26
P-35	21-36	SA-1000	4:57	42	Recycle	12	52	275	1,000	25.0	21.31	64.63	2.75	2.75	4.26	26	26
P-36	21-37	SA-1000	5:00	42	Recycle	12	35	260	3.4/min.	25.0	22.12	68.3	2.50	2.50	3.83	30	30
P-37	21-38	SA-1000	5:55	42	Recycle	12	80	340	7/min.	27.0	15.50	111.63	5.00	5.00	4.48	34	34
P-38	21-39	SA-1000	5:70	42	Recycle	12	65	290	4/min.	26.0	13.22	113.0	12.87	12.87	11.4	34	34
P-39	21-40	SA-1000	4:50	42	Recycle	12	80	350	430	35.0	15.0	100.87	6.87	6.87	8.75	42	42
P-40	21-41	SA-1000	4:52	42	Recycle	12	45	215	220	15.82	17.50	86.56	15.54	15.54	22.1	37	37
P-41	21-42	SA-1000	1:50	42	Recycle	5	182	305	7/min.	10.0	8.75	45.25	15.24	15.24	33.7	37	37
P-42	21-43	SA-1000	4:55	42	Recycle	12	51	240	1,000	16.67	17.46	51.25	4.43	4.43	8.64	26	26
P-43	21-44	SA-1000	3:10	42	Recycle	5	75	210	295	11.0	5.88	23.9	6.61	6.61	27.5	26	26
P-44	21-45	SA-1000	4:76	42	Recycle	9	55	210	575	17.0	13.56	75.55	13.12	13.12	17.4	39	39
P-45	21-46	SA-1000	5:36	42	Recycle	12	50	175	6/min.	24.0	17.50	92.27	12.75	12.75	13.7	37	37
P-46	21-47	SA-1000	5:70	42	Recycle	12	70	280	500	34.0	16.25	106.05	5.0	5.0	4.71	47	47
P-47	21-48	SA-1000	5:21	42	Recycle	12	110	290	7/min.	30.0	16.56	96.15	12.81	12.81	12.6	52	52
P-48	21-49	SA-1000	5:78	42	Recycle	5	100	305	750	15.50	14.0	78.25	5.93	5.93	11.2	34	34
P-49	21-50	SA-1000	4:30	42	Recycle	5	80	30	525	13.0	11.62	73.61	11.62	11.62	15.6	31	31
P-50	21-51	SA-1000	4:72	42	Recycle	5	51	255	500	13.0	14.75	75.54	9.12	9.12	12.0	37	37
P-51	21-52	SA-1000	5:76	42	Recycle	5	100	350	1,000	10.0	15.87	82.4	21.37	21.37	26.0	38	38
P-52	21-53	SA-1000	5:65	42	Recycle	5	65	115	750	11.0	17.0	80.13	21.15	21.15	26.4	46	46
P-53	21-54	SA-1000	5:31	42	Recycle	5	80	210	700	15.0	16.81	81.12	23.04	23.04	28.4	35	35
P-54	21-55	SA-1000	4:55	42	Recycle	5	45	225	310	17.0	16.25	73.25	9.37	9.37	12.6	35	35
P-55	21-56	SA-1000	6:13	42	Recycle	12	45	170	260	20.0	24.0	78.12	9.24	9.24	12.1	46	46
P-56	21-57	SA-1000	6:12	42	Recycle	12	100	300	1,000	20.0	19.56	85.37	9.42	9.42	11.0	43	43
P-57	21-58	SA-1000	6:12	42	Recycle	12	50	160	4/min.	11.0	11.12	45.75	10.0	10.0	21.9	28	28
P-58	21-59	SA-1000	3:52	42	Recycle	5	90	525	600	15.19	10.62	56.13	9.06	9.06	16.3	35	35
P-59	21-60	SA-1000	3:50	42	Recycle	5	90	525	600	15.19	10.62	56.13	9.06	9.06	16.3	35	35

TABLES

TABLE J1 (Continued)

MELTING RECORD

Mater. No.	Mater. Name	MELTING			VACUUM - MICRONS			SKILL WT.			Fidelity Spiral in.	Gross Heat Wt. Lb.		Net Heat Wt. Lb.		Percent Humidity Yrds. Percent	
		Time Min.	Elect. Dia.	Crutible Dia. in.	Start	Melt	After Four	5 Min. Leak Rate	Start	Finish							
1-10	SA-1000	4:54	Recycle	9	75	400	510	2.6/min.	18.75	13.18	None	76.19	19.25	25.3	34	25.3	34
1-11	SA-1000	4:54	Recycle	9	60	155	350	2/min.	13.0	11.62	None	72.61	30.22	41.7	44	41.7	44
1-12	SA-1000	4:54	Recycle	9	65	200	310	10/min.	30.0	11.96	None	6.8	6.8	19.6	31	19.6	31
1-13	SA-1000	4:54	Recycle	9	100	245	300	10/min.	12.0	8.19	None	46.0	6.75	16.6	25	16.6	25
1-14	SA-1000	4:54	Recycle	12	60	150	225	6/min.	22.0	20.31	None	64.62	22.32	28.2	36	28.2	36
1-15	SA-1000	4:54	Recycle	9	55	175	205	4/min.	15.0	11.19	15	70.87	9.24	13.1	48	13.1	48
1-16	SA-1000	4:54	Recycle	12	65	240	375	6/min.	25.0	16.31	15	90.12	9.81	10.6	35	10.6	35
1-17	SA-1000	4:54	Recycle	12	55	245	350	6/min.	22.0	14.06	15	75.31	6.74	8.6	35	8.6	35
1-18	SA-1000	4:54	Recycle	12	70	190	210	7/min.	18.0	15.62	15	64.25	6.62	8.0	42	8.0	42
1-19	SA-1000	4:54	Recycle	12	50	250	275	5/min.	25.0	16.96	None	80.81	26.40	25.2	41	25.2	41
1-20	SA-1000	4:54	Recycle	12	75	300	550	5.2/min.	25.25	26.25	15	105.50	27.56	29.2	37	29.2	37
1-21	SA-1000	4:54	Recycle	12	100	360	750	10/min.	30.0	16.62	21	110.06	37.00	3.1	45	3.1	45
1-22	SA-1000	4:54	Recycle	12	150	400	750	10/min.	35.0	20.0	16	123.37	32.17	26.6	47	26.6	47
1-23	SA-1000	4:54	Recycle	12	55	310	650	4/min.	35.0	14.31	None	60.68	12.75	12.7	43	12.7	43
1-24	SA-1000	4:54	Recycle	9	65	290	350	5/min.	23.0	23.12	14	124.62	12.75	10.2	46	10.2	46
1-25	SA-1000	4:54	Recycle	9	75	200	400	6/min.	20.0	23.0	None	75.67	11.11	14.7	46	14.7	46
1-26	SA-1000	4:54	Recycle	12	55	210	500	5/min.	17.0	18.75	15	75.67	9.12	14.7	32	14.7	32
1-27	SA-1000	4:54	Recycle	12	55	240	430	7/min.	20.0	22.67	14	62.31	9.12	14.7	32	14.7	32
1-28	SA-1000	4:54	Recycle	9	75	250	1,000	7/min.	20.0	11.0	None	61.62	12.25	10.2	30	10.2	30
1-29	SA-1000	4:54	Recycle	9	75	250	1,000	7/min.	20.0	3.67	14	65.50	9.00	13.0	40	13.0	40
1-30	SA-1000	4:54	Recycle	12	400	225	400	5/min.	25.0	16.86	17	110.50	25.06	24.6	44	24.6	44
1-31	SA-1000	4:54	Recycle	12	150	450	525	7/min.	35.0	22.36	15	120.66	22.50	21.9	54	21.9	54
1-32	SA-1000	4:54	Recycle	12	40	575	450	6/min.	30.0	16.06	15	63.12	5.93	10.7	40	10.7	40
1-33	SA-1000	4:54	Recycle	12	100	245	430	4/min.	25.0	15.31	17	91.12	15.11	16.6	42	16.6	42
1-34	SA-1000	4:54	Recycle	12	100	275	750	5/min.	33.0	13.6	15	104.13	14.77	14.1	40	14.1	40
1-35	SA-1000	4:54	Recycle	12	65	250	405	5/min.	25.0	21.25	23	140.44	24.50	10.7	39	10.7	39
1-36	SA-1000	4:54	Recycle	12	55	215	375	7/min.	35.0	32.0	None	135.12	43.96	7.2	36	7.2	36
1-37	SA-1000	4:54	Recycle	12	55	250	600	7/min.	42.0	24.31	20	155.38	24.16	16.5	40	16.5	40
1-38	SA-1000	4:54	Recycle	12	70	300	625	5/min.	32.0	24.06	15	144.50	24.43	16.9	35	16.9	35
1-39	SA-1000	4:54	Recycle	12	50	325	625	11/min.	32.0	24.36	16	155.37	24.25	15.6	36	15.6	36
1-40	SA-1000	4:54	Recycle	12	35	250	700	3/min.	25.0	25.25	16	145.63	24.32	16.2	47	16.2	47
1-41	SA-1000	4:54	Recycle	12	65	260	700	5/min.	25.0	25.36	15	145.75	24.16	16.5	46	16.5	46
1-42	SA-1000	4:54	Recycle	12	5	250	1,000	2.4/min.	25.0	27.63	17	145.63	24.16	16.5	51	16.5	51
1-43	SA-1000	4:54	Recycle	12	40	325	600	7/min.	28.25	46.75	15	131.12	24.10	16.3	53	16.3	53
1-44	SA-1000	4:54	Recycle	12	55	340	1,000	2/min.	37.65	22.50	20	166.50	24.60	14.7	54	14.7	54
1-45	SA-1000	4:54	Recycle	12	75	290	325	5/min.	24.25	24.0	16	101.12	26.60	26.0	35	26.0	35
1-46	SA-1000	4:54	Recycle	12	100	275	700	8/min.	25.0	21.87	12	62.16	9.12	11.0	43	11.0	43
1-47	SA-1000	4:54	Recycle	12	45	240	355	5/min.	25.6	17.06	14	77.25	9.26	11.9	45	11.9	45
1-48	SA-1000	4:54	Recycle	12	65	340	395	5/min.	22.0	6.69	11	62.16	9.00	14.4	37	14.4	37
1-49	SA-1000	4:54	Recycle	12	55	300	365	6/min.	19.00	17.37	None	52.06	9.00	14.5	36	14.5	36

TABLES

TABLE J1 (Continued)

MELTING RECORD

Heat No.	Order No.	Material	MELTING			VACUUM - MICRONS			SKULL WT.		Fluidity Spiral in.	Gross Heat Wt. Lb.		Net Heat Wt. Lb.		Percent Humidity Yield Percent	
			Time Min.	Volt	Amperes	Start	During	After	Start	Finish							
P-70	22-02	DAI-mw	5.70	35	14,000	9	60	290	400	1/min.	10.00	14.37	68.19	9.00	13.1	40	
P-71	22-02	DAI-mw	4.42	41	14,000	12	65	365	305	5/min.	25.50	17.31	77.56	20.91	27.0	44	
P-72	22-02	C.P.T.	4.60	42	12,500	12	100	245	700	13/min.	53.00	21.25	130.75	9.67	7.3	45	
P-73	22-02	DAI-mw	2.84	40	12,000	5	48	290	330	1.2/min.	15.00	7.44	43.88	7.61	17.2	37	
P-74	22-02	DAI-mw	1.45	35	15,000	5	120	295	600	5/min.	11.00	7.37	17.88	5.87	32.5	42	
P-75	22-02	DAI-mw	6.30	42	14,000	12	85	305	730	9/min.	25.00	21.75	108.75	25.62	23.6	49	
P-76	22-02	DAI-mw	5.45	40	14,000	12	45	230	625	6/min.	26.00	26.25	56.75	27.50	28.5	50	
P-77	22-02	Cancelled	5.65	40	11,000	9	70	145	425	2/min.	12.00	15.31	79.37	9.13	11.5	38	
P-78	22-02	DAI-mw	5.21	40	12,000	9	45	165	330	7/min.	15.00	13.75	76.50	16.86	22.3	50	
P-79	22-02	DAI-mw	5.60	42	14,000	12	100	270	500	7/min.	22.38	27.50	94.56	8.18	5.6	47	
P-80	22-02	DAI-mw	5.45	40	12,000	9	45	175	240	3/min.	17.50	12.50	79.00	21.22	26.5	40	
P-81	22-02	SH-20-24	3.60	36	14,000	9	57	175	240	4/min.	8.00	11.38	52.68	9.12	17.2	47	
P-82	22-02	DAI-mw	3.75	40	14,000	9	37	185	250	7/min.	13.50	10.88	61.62	9.18	14.8	38	
P-83	22-02	DAI-mw	5.25	41	14,000	12	46	256	230	1.6/min.	32.44	24.56	83.69	20.53	24.5	53	
P-84	22-02	DAI-mw	4.00	41	14,000	12	45	140	430	3/min.	18.56	14.60	74.87	14.97	19.9	49	
P-85	22-02	DAI-mw	4.40	40	14,000	12	45	170	750	1/min.	21.50	10.25	63.00	31.66	38.0	44	
P-86	22-02	DAI-mw	1.50	35	13,000	5	50	200	290	4/min.	9.44	9.00	20.16	5.87	29.0	53	
P-87	22-02	DAI-mw	4.90	41	12,000	5	55	245	900	1/min.	15.00	14.31	81.25	17.18	21.1	46	
P-88	22-02	DAI-mw	6.30	41	12,000	9	65	275	900	7/min.	20.00	15.50	93.66	17.10	18.2	51	
P-89	22-02	DAI-mw	5.20	41	14,000	9	75	290	900	7/min.	21.00	14.87	94.56	9.12	9.6	47	
P-90	22-02	DAI-mw	4.80	42	14,000	12	38	22-295	750	1.4/min.	30.00	15.00	52.00	8.99	9.7	53	
P-91	22-02	DAI-mw	6.30	35	12,000	9	60	255	400	7/min.	15.00	20.25	93.69	11.62	12.4	51	
P-92	22-02	DAI-mw	4.85	40	14,000	12	38	240	450	4/min.	35.00	23.75	102.12	11.99	11.7	42	
P-93	22-02	DAI-mw	7.05	41	14,000	9	45	205	1,000	11/min.	14.00	12.87	97.38	13.16	13.5	54	
P-94	22-02	DAI-mw	3.15	40	14,000	9	37	500	245	2.8/min.	30.00	94.64	82.68	22.74	27.5	48	
P-95	22-02	DAI-mw	4.55	40	14,000	12	70	210	245	7/min.	16.00	16.00	47.25	12.35	26.1	44	
P-96	22-02	DAI-mw	3.67	40	12,000	9	45	340	625	7/min.	11.00	8.50	76.37	22.74	29.7	53	
P-97	22-02	DAI-mw	4.90	43	12,000	9	65	290	345	5/min.	17.00	11.15	77.37	22.91	29.5	51	
P-98	22-02	DAI-mw	5.41	42	12,000	9	45	335	390	3/min.	15.00	12.75	77.12	23.47	30.4	46	
P-99	22-02	DAI-mw	4.77	40	12,000	9	32	175	255	2.4/min.	18.00	13.94	76.50	23.03	30.1	44	
P-100	22-02	DAI-mw	4.80	42	12,000	9	42	240	305	3/min.	24.56	17.12	80.00	23.10	28.8	42	
P-101	22-02	DAI-mw	4.65	42	14,000	9	85	220-290	345	7/min.	15.50	11.25	46.44	7.43	15.0	35	
P-102	22-02	DAI-mw	2.90	40	12,000	5	35	310	700	9/min.	25.00	11.63	54.12	9.06	14.8	45	
P-103	22-02	DAI-mw	3.25	37	14,000	9	35	145	290	13/min.	11.06	9.36	85.56	30.72	35.8	55	
P-104	22-02	DAI-mw	5.10	42	12,000	9	32	300	1,000	1.4/min.	30.00	18.06	61.12	12.05	19.7	57	
P-105	22-02	DAI-mw	3.75	41	12,000	9	65	240	325	2/min.	21.00	10.50	51.00	4.82	9.4	53	
P-106	22-02	C.P.T.	3.35	40	14,000	9	57	32-98	102	16/min.	11.38	13.12	50.62	5.16	10.1	46	
P-107	22-02	C.P.T.	3.67	7	14,000	9	37	15-89	120	2/min.	11.62	13.00	49.68	5.06	10.1	45	
P-108	22-02	C.P.T.	3.45	40	10,500	9	40	70	125	2/min.	10.25	15.37					

TABLES

TABLE J1 (Continued)

MELTING RECORD

Order No.	MATERIAL	MELTING			VACUUM - MICRONS			SKULL WT.			FLUIDITY			GROSS			NET			PERCENT HUMIDITY		
		Time	Elect.	Crucible	Start	Stop	Leak Rate	Start	Stop	Leak Rate	Spinal	Heat	Wt.	Heat	Wt.	Yield	Heat	Wt.	Yield	Percent	Humidity	
		Water	Wt.	Vol.	Wt.	Vol.	Wt.	Wt.	Wt.	Wt.	Wt.	Wt.	Wt.	Wt.	Wt.	Wt.	Wt.	Wt.	Wt.	Wt.	Wt.	
1000	1000	3.44	30	9	30	22-85	180	2/min.	10.50	13.31	6	45.16	4.61	45.16	4.61	9.6	45.16	4.61	9.6	52	52	
1000	1000	3.44	30	9	45	22-65	98	2/min.	10.44	12.62	9	61.06	4.93	61.06	4.93	9.6	61.06	4.93	9.6	52	52	
1000	1000	3.44	30	9	41	22-31	102	1.6/min.	10.00	12.81	None	51.00	4.61	51.00	4.61	9.6	51.00	4.61	9.6	56	56	
1000	1000	3.44	30	9	45	22-150	180	6/min.	11.75	12.81	6	50.25	5.12	50.25	5.12	10.1	50.25	5.12	10.1	54	54	
1000	1000	3.44	30	9	33	13-105	180	2/min.	6.21	17.37	11	47.56	4.81	47.56	4.81	10.1	47.56	4.81	10.1	36	36	
1000	1000	3.44	30	9	72	125	195	5/min.	5.00	8.50	62	46.43	5.93	46.43	5.93	20.5	46.43	5.93	20.5	52	52	
1000	1000	3.44	30	9	40	22-90	195	2/min.	10.75	12.45	8	51.12	9.12	51.12	9.12	17.0	51.12	9.12	17.0	57	57	
1000	1000	3.44	30	9	44	22-90	650	2/min.	25.00	11.50	14	77.00	6.81	77.00	6.81	6.7	77.00	6.81	6.7	48	48	
1000	1000	3.44	30	9	95	390	900	5/min.	30.00	15.87	15	55.50	11.62	55.50	11.62	12.1	55.50	11.62	12.1	44	44	
1000	1000	3.44	30	9	125	495	550	37/min.	20.00	13.75	None	53.16	7.90	53.16	7.90	14.6	53.16	7.90	14.6	44	44	
1000	1000	3.44	30	9	55	473	750	16/min.	20.00	25.25	14	142.55	5.30	142.55	5.30	3.2	142.55	5.30	3.2	40	40	
1000	1000	3.44	30	9	40	473	750	2/min.	30.00	26.75	15	108.12	30.65	108.12	30.65	25.0	108.12	30.65	25.0	35	35	
1000	1000	3.44	30	9	49	34-250	650	14/min.	35.50	12.00	14	75.37	5.16	75.37	5.16	12.1	75.37	5.16	12.1	43	43	
1000	1000	3.44	30	9	65	395	600	3/min.	35.00	21.61	11	113.12	26.37	113.12	26.37	23.3	113.12	26.37	23.3	52	52	
1000	1000	3.44	30	9	105	390	600	7/min.	22.37	19.00	13	95.75	13.93	95.75	13.93	13.5	95.75	13.93	13.5	57	57	
1000	1000	3.44	30	9	45	195	450	4/min.	20.00	14.50	None	50.45	7.25	50.45	7.25	14.3	50.45	7.25	14.3	52	52	
1000	1000	3.44	30	9	53	455	495	3/min.	20.00	15.50	None	57.81	6.61	57.81	6.61	11.7	57.81	6.61	11.7	42	42	
1000	1000	3.44	30	9	125	200	350	10/min.	60.45	4.75	None	55.30	7.31	55.30	7.31	15.3	55.30	7.31	15.3	45	45	
1000	1000	3.44	30	9	100	200	350	8/min.	20.00	14.50	None	56.66	7.37	56.66	7.37	13.5	56.66	7.37	13.5	46	46	
1000	1000	3.44	30	9	35	350	300	15/min.	16.25	9.30	102	54.14	12.24	54.14	12.24	22.6	54.14	12.24	22.6	52	52	
1000	1000	3.44	30	9	35	350	300	3.6/min.	16.00	69.00	None	105.06	20.61	105.06	20.61	19.6	105.06	20.61	19.6	43	43	
1000	1000	3.44	30	9	70	200-750	600	16/min.	16.00	18.12	None	57.67	11.61	57.67	11.61	11.6	57.67	11.61	11.6	55	55	
1000	1000	3.44	30	9	44	325	550	2.6/min.	22.00	26.00	1	64.12	4.50	64.12	4.50	7.0	64.12	4.50	7.0	50	50	
1000	1000	3.44	30	9	100	250	550	6/min.	21.00	13.64	None	85.44	5.24	85.44	5.24	10.6	85.44	5.24	10.6	54	54	
1000	1000	3.44	30	9	25	100	200	10/min.	10.00	5.16	None	21.66	5.61	21.66	5.61	26.7	21.66	5.61	26.7	46	46	
1000	1000	3.44	30	9	50	150-275	500	10/min.	23.00	15.12	None	74.12	4.62	74.12	4.62	5.2	74.12	4.62	5.2	40	40	
1000	1000	3.44	30	9	50	150-300	500	7/min.	26.00	19.67	None	65.60	4.62	65.60	4.62	7.0	65.60	4.62	7.0	37	37	
1000	1000	3.44	30	9	80	200-350	750	3.6/min.	17.00	49.75	None	77.56	4.62	77.56	4.62	5.9	77.56	4.62	5.9	46	46	
1000	1000	3.44	30	9	35	300-500	1000	4/min.	18.00	17.50	None	65.06	11.36	65.06	11.36	16.4	65.06	11.36	16.4	46	46	
1000	1000	3.44	30	9	20	15	60	3/min.	9.25	9.67	121	50.16	10.10	50.16	10.10	20.1	50.16	10.10	20.1	52	52	
1000	1000	3.44	30	9	65	200-350	500	10/min.	16.06	17.25	None	50.81	4.62	50.81	4.62	9.0	50.81	4.62	9.0	44	44	
1000	1000	3.44	30	9	70	150-325	600	7/min.	18.00	12.62	None	52.12	4.74	52.12	4.74	9.0	52.12	4.74	9.0	46	46	
1000	1000	3.44	30	9	20	300-750	525	5/min.	23.00	12.93	None	31.81	10.00	31.81	10.00	31.4	31.81	10.00	31.4	47	47	
1000	1000	3.44	30	9	60	250-400	500	6/min.	18.00	10.50	None	55.50	4.75	55.50	4.75	8.0	55.50	4.75	8.0	38	38	
1000	1000	3.44	30	9	25	300-500	1000	6/min.	18.00	14.62	None	75.44	9.84	75.44	9.84	13.0	75.44	9.84	13.0	46	46	

TABLE J1 (Continued)

MELTING RECORD

Heat No.	Order No.	Material	MELTING			VACUUM - MICRONS			SKULL WT.		Fluidity Spiral In.	Gross Heat Wt.		Net Heat Wt.		Percent Humidity	
			Time Min.	Volts	Amperes	Time Min.	Start	After 5 Min. Leak Rate	Start	Finish		Lb.	Lb.	Lb.	Lb.	Yield	Percent
P552	23-80	6A1-m	2:16	41	12,500	Recycle	9	325	10:50	5:87	None	27.61	-	36	-	1.3	43
P553	23-81	6A1-m	3:10	44	14,000	Recycle	12	350	18:00	14:18	None	51.31	-	5.36	-	11.2	36
P554	23-82	6A1-m	3:54	44	14,000	Recycle	12	750	23:87	13:37	None	50.00	-	5.30	-	10.6	44
P555	23-83	6A1-m	5:25	44	14,000	Recycle	12	750	18:00	21:50	None	103.88	-	21.10	-	20.3	41
P556	23-84	6A1-m	5:56	44	14,000	Recycle	12	420	18:00	12:87	None	63.37	-	5.22	-	6.2	49
P557	23-85	6A1-m	5:52	44	14,000	Recycle	12	750	18:00	18:00	None	109.50	-	9.84	-	8.9	46
P558	23-86	6A1-m	5:15	44	15,000	Recycle	12	250	18:00	15:62	None	66.56	-	14.35	-	16.0	30
P559	23-87	6A1-m	12:12	44	16,000	Recycle	16	350	70:00	56:25	None	277.00	-	74.00	-	26.0	48
P560	23-88	6A1-m	5:12	44	14,000	Recycle	12	600	18:00	15:75	None	95.06	-	11.22	-	11.0	38
P561	23-89	6A1-m	5	44	14,000	Recycle	12	750	18:00	17:64	None	90.3	-	10.50	-	10.0	49
P562	23-90	6A1-m	3:25	44	14,000	Recycle	9	250	10:00	6:37	None	37.52	-	-	-	-	62
P563	23-91	6A1-m	4:40	44	14,500	Recycle	7	500	20:00	16:16	None	80.00	-	5.48	-	6.80	60
P564	23-92	6A1-m	5:45	44	14,000	Recy. 5-6	12	500	18:75	17:00	None	85.25	-	10.86	-	12.0	48
P565	23-93	6A1-m	7:24	44	15,000	Recy. 7-6	12	250	27:00	27:51	None	146.67	-	44.66	-	30.0	55
P566	23-94	6A1-m	4:52	44	14,500	Recycle 6	12	550	20:00	14:25	None	60.62	-	-	-	-	46
P567	23-95	6A1-m	5:11	44	14,000	Recycle 7	12	200	18:00	14:62	None	74.67	-	10.66	-	14.0	49
P568	23-96	6A1-m	5:04	44	14,000	Recycle 7	12	300	18:00	15:00	None	62.37	-	-	-	-	46
P569	23-97	6A1-m	5:27	44	15,000	Recycle 7	12	500	20:00	15:50	None	75.00	-	10.66	-	13.0	46
P570	23-98	6A1-m	5:30	44	15,000	Recycle 7	12	500	18:00	16:70	None	91.37	-	14.11	-	15.0	55
P571	23-99	6A1-m	5:34	44	15,000	Recycle 7	12	80	20:00	15:12	None	60.65	-	10.54	-	17.0	55
P572	23-100	6A1-m	5:60	44	14,000	Recycle 7	12	175	19:75	15:00	None	61.65	-	-	-	-	52
P573	23-101	6A1-m	7:50	44	15,000	3-Sq. Blks.	12	500	30:00	27:62	None	142.00	-	22.61	-	15.0	50
P574	23-102	6A1-m	4:05	44	15,000	3 Chunks	12	500	20:00	18:37	None	69.50	-	14.21	-	15.0	55
P575	23-103	6A1-m	2:20	44	15,000	Recycle 4	9	350	20:75	9:75	None	42.00	-	11.56	-	27.0	47
P576	23-104	6A1-m	6:05	44	15,000	7 63-Bars	12	750	20:00	16:32	None	68.52	-	13.76	-	15.0	46
P577	23-105	6A1-m	7:55	44	15,000	7 50-Sq. Blks.	12	1,000	20:00	26:50	None	135.12	-	36.36	-	26.0	-
P578	23-106	6A1-m	4:55	44	15,000	6 619g	12	750	20:00	18:37	None	88.00	-	15.46	-	17.0	46
P579	23-107	6A1-m	4:55	44	14,500	6 619g	12	350	40:00	50:00	None	73.75	-	13.96	-	16.0	47
P580	23-108	6A1-m	4:76	44	15,000	7 Square	12	200	20:00	17:87	None	140.00	-	41.24	-	25.0	57
P581	23-109	6A1-m	7:44	44	15,000	Recycle	12	250	20:00	26:12	6	106.75	-	16.34	-	15.0	66
P582	23-110	6A1-m	6:22	44	15,000	6 619g	12	750	20:00	19:50	6	64.67	-	9.86	-	15.0	56
P583	23-111	6A1-m	3:55	44	14,000	4-Recy.	12	850	25:30	13:00	6	118.25	-	23.24	-	19.0	60
P584	23-112	6A1-m	6:05	44	15,000	Recycle	12	1,000	25:00	22:31	6	145.00	-	22.20	-	15.0	54
P585	23-113	6A1-m	7:55	44	14,000	7-Recy.	12	350	16:12	21:25	6	53.75	-	14.32	-	15.0	60
P586	23-114	6A1-m	5:75	44	14,000	6-Recy.	12	1,000	42:00	44:12	6	269.00	-	45.36	-	16.1	62
P587	23-115	6A1-m	12:10	44	16,000	7-Recy.	16	350	22:87	19:00	6	102.81	-	16.62	-	10.0	74
P588	23-116	6A1-m	5:75	44	13,000	Recycle	12	1,000	27:48	16:00	-	100.00	-	-	-	-	55
P589	23-117	6A1-m	6:05	44	13,500	7-Recy.	12	400	32:63	34:31	6	160.25	-	36.52	-	22.0	-
P590	23-118	6A1-m	6:50	44	13,500	6-Recy.	12	350	23:93	23:12	6	116.65	-	11.62	-	65.0	34

TABLE J1 (Continued)

MELTING RECORD

Heat No.	Proe	Meter	MELTING			VACUUM - MICRONS			SKULL WT.		Fluidity Spiral	Gross Heat Wt.		Net Heat Wt.		Municipity % C	
			Time Min.	Elect. Dia.	Crucible Dia.	Start	Melt	After	2 Min. Lead Rate	End		Heat Wt. Lb.	Skull Wt. Lb.	Heat Wt. Lb.	Skull Wt. Lb.		
822	4-110	55-100	40-45	7	Recy.	5	100	150-200	245	4/m c.	11.00	6.50	2	49.31	10.46	21.0	35
823	4-110	55-100	-	-	7	Recy.	5	40	-	-	-	-	-	-	-	-	-
824	4-110	55-100	40-45	7	Recy.	12	100	210-300	550	2/m c.	33.66	15.37	4	103.6	11.66	11.0	41
825	4-110	55-100	40-45	7	Recy.	12	30	40-100	500	5/m c.	26.75	15.00	6	98.00	35.36	40.0	31
826	4-110	55-100	40-45	7	Recycle	12	30	150-300	600	2/m c.	24.50	15.56	6	51.96	14.62	15.0	
827	4-110	55-100	40-45	7	Recy.	12	20	100-350	750	2/m c.	35.50	45.00	-	162.50	73.75	24.0	42
828	4-110	55-100	40-45	7	Recy.	12	20	100-350	500	2/m c.	37.45	35.42	-	184.50	21.24	7.5	35
829	4-110	55-100	40-45	7	Recy.	12	30	100-350	500	5/m c.	43.0	45.0	-	220.00	111.2	21.0	45
830	4-110	55-100	40-45	7	Recy.	12	70	500-700	1,000	7/m c.	30.0	46.0	-	252.00	51.62	24.0	42
831	4-110	55-100	40-45	7	Recy.	12	5	100-250	750	3/m c.	16.25	35.0	-	197.00	34.56	16.0	37
832	4-110	55-100	40-45	7	Recy.	5	30	150	450	4/m c.	6.66	11.61	10	53.25	12.0	24.0	45
833	4-110	55-100	40-45	7	Recy.	5	125	125-150	150	5/m c.	7.45	10.0	10.5	55.0	12.0	24.0	45
834	4-110	55-100	40-45	7	Recy.	5	10	-	-	10/m c.	7.0	5.0	-	323.0	44.42	3.0	36
835	4-110	55-100	40-45	7	Recy.	5	300	750	1,000	10/m c.	53.0	55.0	-	323.0	44.42	13.0	50
836	4-110	55-100	40-45	7	Recy.	5	200	200-250	1,000	5/m c.	45.31	55.25	-	323.0	44.42	13.0	50
837	4-110	55-100	40-45	7	Recy.	5	225	200-250	1,000	5/m c.	74.34	55.43	-	323.0	44.42	13.0	50
838	4-110	55-100	40-45	7	Recy.	5	225	200-250	1,000	5/m c.	41.50	48.13	-	290.0	64.24	2.0	55
839	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
840	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
841	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
842	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
843	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
844	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
845	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
846	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
847	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
848	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
849	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
850	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
851	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
852	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
853	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
854	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
855	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
856	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
857	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
858	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
859	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
860	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
861	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
862	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
863	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
864	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
865	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
866	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
867	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
868	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
869	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
870	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
871	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
872	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
873	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
874	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
875	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
876	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
877	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
878	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
879	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
880	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
881	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
882	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
883	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
884	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
885	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
886	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
887	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
888	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
889	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
890	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
891	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
892	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
893	4-110	55-100	40-45	7	Recy.	5	250	300	1,000	5/m c.	61.32	48.0	-	323.0	44.42	13.0	50
894	4-110	55-100	40														

TABLE J2

MECHANICAL PROPERTIES OF
RECYCLED LOW INTERSTITIAL 6Al-4V

HEAT NO.		YIELD STRENGTH 0.2% OFFSET	ULTIMATE TENSILE STRENGTH	ELONGATION	REDUCTION IN AREA %	BHN
22-43-P245-	1A	104,000	124,800	13	33.6	269.3
	B	103,000	124,000	10	24.6	
	C	105,000	124,200	9	22.4	
22-43-P249-	1A	102,000	122,200	10	19.0	269.3
	B	104,000	124,200	10	21.7	
	C	104,000	124,000	10	24.6	
22-43-P252-	2A	106,000	127,600	10	20.4	276.5
	B	107,000	127,600	11	25.2	
	C	104,000	126,000	9	22.4	
22-43-P255-	3A	107,000	128,000	13	27.2	277.0
	B	107,000	126,800	10	25.9	
	C	106,000	126,800	10	21.7	
22-43-P259-	1A	109,000	128,000	13	32.4	279.6
	C	108,000	128,000	12	24.4	
22-43-P289-	5A	106,000	126,000	12	31.9	277.0
	B	104,000	122,000	9	20.8	
	C	105,000	122,000	12	31.9	
22-43-P267-	2A	109,000	130,000	12	26.5	271.6
	B	110,000	131,200	12	23.9	
	C	110,000	130,000	10	19.7	
22-43-P271-	1A	109,000	128,000	13	29.2	279.0
	B	110,000	128,400	12	26.5	
	C	108,000	128,000	10	24.6	
22-43-P280-	2A	110,000	128,000	8	22.4	277.0
	B	110,000	130,000	11	27.0	
	C	109,000	130,000	13	34.4	
22-43-P282-	4A	Too Brittle	93,200	0	0	282.3
	B		96,000	0	0	
	C		72,000	0	0	
22-43-P332-C1A		105,000	125,200	12	18.1	272.0
	B	102,000	123,600	12	33.6	
	C	104,000	124,000	12	27.0	

TABLE J2 (Continued)

MECHANICAL PROPERTIES OF
RECYCLED LOW INTERSTITIAL 6Al-4V

HEAT NO.	YIELD STRENGTH 0.2% OFFSET	ULTIMATE TENSILE STRENGTH	ELONGATION	REDUCTION IN AREA %	BHN
22-43-P334-C2A	109,000	128,800	10	28.5	290.1
B	110,000	128,800	11	27.7	
C	110,000	128,000	11	29.2	
22-43-P372-C3A	110,000	131,600	11	26.5	279.2
B	110,000	131,600	12	23.2	
C	112,000	131,200	11	26.5	
22-43-P375-C4A	114,000	133,200	8	15.0	287.2
B	116,000	134,400	10	19.7	
C	115,000	134,800	11	26.5	
22-43-P376-C5A	117,000	137,200	13	26.5	296.0
B	119,000	136,200	10	21.0	
C	118,000	136,200	11	21.0	

TABLE J3

EXPERIMENTAL RAMMED GRAPHITE MOLD MIXTURES

MIX NO.	GRAPHITE USED			PITCH	LAUNDRY STARCH	CORN FLOUR	CEMENT C-3	RAW LINSEED OIL	WATER
	BB5	Reclaim BB5	Classified Turnings						
1	69			10	5		8		8
2	70.5			10	5.25		8.25		6
3		80.5		5.8	5.8		4.6		3.3
4		75		7.0	5.0		6.0		7.0
5		70		10	5.0		8.0		7.0
6		73.7		10.1		3.04	8.1		5.05
7	83.7				7.1				10.1
8	76.1			11					12.9
9	79.3						9.2		11.5
10	73.5			10.6	5.32				10.6
11	71.1			10.3			8.3		10.3
12	85.8					3.73			10.4
13	75			10.85		3.26			10.85
14		86.25				3.75			10.0
15		86.25				(Truscar) 3.75			10.0
16		70.5*		10	5.25		8.25		6.0
17	70			10	5		8		7
18	73			12.2	6.1				8.7
19	77			10	5			1	7
20	74.76			9.85	4.92			1.97	8.5
21	72			10	5			3	10
22		70 (BBB)		10	5		8		7
23	74			10	5			4	7
24	76			10		3		1	10
25	75			10		3		2	10
26	74			10		3		3	10
27	75			10		4		1	10
28	74			10		4		2	10
29	73			10		4		3	10
30			70	10	5		8		7
31			70	10		3	8		9
32			67.75	9.67	4.84		7.75		10
33			68	10	5		8		9
34			69	10		4	7		10
35			70	10		4	6		10
36			71	10		4	5		10
37			72	10		4	4		10
38			73	10		4	3		10

* 2nd reclaim

TABLE J3 (Continued)

EXPERIMENTAL RAMMED GRAPHITE MOLD MIXTURES

MIX NO.	GRAPHITE USED			DUPONOL G	PITCH	LAUNDRY STARCH	CORN FLOUR	CEMENT C-3	RAW LINSEED OIL	WATER
	BB5	Crescent	Classified Turnings							
39		65		0.938	9.38		2.82	7.5		14.75
40		70			10		3	8		9
41			69	1	10		3	8		9
42			69		10	5		8		8
43			73		10		4	2		11
44			66.2	1.09	10.95	5.47		8.75		7.65
45			68	2	10	5		8		7
46		70			10	5		8		7
47	70				10	5		8		7
48	69.5		70		10	5		8		7
49	72.5			0.98	9.72	4.85		2.78		8.35
50	69			1	9.8	4.9			2.94	8.82
					10	5		8		7
			(Calcined Coke 3-15)							
51	70		70		10	5		8		7
52					9.72	4.86		3.87		9.58
			(Calcined Coke 3-04)							
			70		10	5		8		7
53			(Calcined Coke 3-15)							
			76							
54	71				8	4		6		6
55	72				10	4		8		7
56	73				10	3		8		7
57	70				10			8		9
58	70				10		3	8		9
59		63.75		0.938	9.38	2.82	7.5			9.38
60			66.5		9.38	2.82	5.62			14.7

TABLE J3 (Continued)

EXPERIMENTAL RAMMED GRAPHITE MOLD MIXTURES

MIX NO.	PETROLEUM CALCINED COKE	COAL CALCINED COKE	PETROLEUM RAW COKE	COKE FLOUR	STARCH	PITCH	C-1 CEMENT	WATER	SHRINKAGE In./In.
83		75		5	5	10*		10	0.006
84		70		10		10*		10	0.007
85		65		15		10*		10	0.002
86		70		5	5	10*		10	0.012
87		65		10	5	10*		10	0.012
88		60		15	5	10*		10	0.011
89		65		5	5	10*	8	7	0.016
90		60		10	5	10*	8	7	0.015
91		55		15	5	10*	8	7	0.013
92			75	5		10*		10	0.105
93			70	10		10*		10	0.089
94			65	15		10*		10	0.074
95			75		5	10*		10	0.097
96			73.8		5.2	10.5*		10.5	0.105
97			72.2		5.56	11.2*		11.2	0.093
98			75		5	10*		10	0.093
99			73.8		5.2	10.5*		10.5	0.093
100			72.2		5.56	11.2*		11.2	0.159
101				80		10*		10	0.002
102				75		15*		10	-----
103				70		20*		10	-----
104				76.2	4.76	9.52*		9.52	0.023
105				71.6	4.7	14.2*		9.3	0.015
106				66.7	4.77	10.03*		9.3	0.021
107		78				12*		10	0.007
108		80				10*		10	0.008
109		82				8*		10	0.003
110	75			5		10*		10	-----
110A	70			5	5	10*		10	0.019
110B	65			10	5	10*		10	0.016
111	70			10		10*		10	0.002
112	65			15		10*		10	0.005
113		76				12**		12	0.016
114		78				10**		12	0.010
115		80				8**		12	0.014
116		71			5	12**		12	0.018
117		73			5	10**		12	0.012
118		75			5	8**		12	0.016
119		76				12***		12	0.009
120		78				10***		12	0.001
121		80				8***		12	-----
122				78		10***		12	0.003
123				73		15***		12	0.007
124				68		20***		12	0.021
125				73	5	10***		12	0.016
126				68	5	15***		12	0.021
127				63	5	20***		12	0.023
128		71			5	12***		12	0.016
129		73			5	10***		12	0.012
130		75			5	8***		12	0.014
131		67			5	12***	8	8	0.016
132		69			5	10***	8	8	0.016
133		71			5	8***	8	8	0.018

* Foundry Pitch

** Black Diamond Pitch

*** Koppers Pitch

TABLE J3 (Continued)

EXPERIMENTAL RAMMED GRAPHITE MOLD MIXTURES

MIX NO.	GRAPHITE USED		PITCH	LAUNDRY STARCH	CEMENT C-3	LIQUID BINDER	WATER	SHRINKAGE In/In.
	Sweco 20	Sweco 40						
61	70		9	6		10 (7605)	5	
62	75		9	6		8 (7605)	2	0.0105
63	70.5		10.1	5.04		10.1 (7601)	3.5	0.0150
64	71.6		10.2	5.1		10.2 (7101)	1.5	0.0276
65	52	21	10	5	8		5.6	0.0239
66	52	20	10	3	8		7	
67	70 (carbon sand)		10	5	8		7	
COKE 93-04)								
68	68		12	5	8	7		0.0094
69	70		10	5	8	7		0.0094
70	72		8	5	8	7		0.0063
71	64.8		11.4	4.75	4.75		14.3	0
72	66.7		9.5	4.75	4.75		14.3	0
73	68.6		7.6	4.75	4.75		14.3	0
74	74.3		11.42				14.28	0
75	75.2		10.50				14.3	0.031
76	78		7.6				14.3	0
77	73		12	5		10		0.0125
78	75		10	5		10		0.0019
79	77		8	5		10		0.0078
80	78		12			10		0.0016
81	80		10			10		0
82	82		8			10		0

* Slurry - Petroleum residual

7605 - Amino - Aldehyde

7601 - Urea type - water soluble

7101 - Pheno - Formaldehyde - water dispersing

TABLE J4

GRAIN SIZE DISTRIBUTIONS OF GRAPHITE BASE MATERIALS

U.S. Series Equivalent No.	National Carbon Grade BB 8	Sweco Separator Pass No. 10	Reclaimed (Tumbled Only)	Reclaimed Crushed & Tumbled	Reclaimed Crushed Only	Reclaimed Crushed & Pulverized	2nd Reclaim Crushed	Reclaim Plus 20% Fines
6								
12		0.78						
20	5.44	39.70	0.12	0.26	0.22	2.32	0.32	0.38
30	40.82	40.86	1.46	3.88	5.54	7.10	5.24	1.50
40	32.35	10.62	5.30	13.50	9.88	12.12	15.06	3.10
50	16.72	3.24	31.30	25.96	20.76	20.06	30.28	15.50
70	2.08	1.68	38.75	29.10	26.20	23.56	25.40	23.78
100	.49	1.00	19.72	18.32	18.10	15.66	10.94	17.60
140	.28	0.52	2.90	5.44	6.86	5.14	3.90	6.50
200	.20	0.46	0.84	2.18	3.80	2.96	2.76	4.30
270	.08	0.14	0.08	0.34	2.00	0.96	1.28	3.30
Pan	.22	0.42	0.06	0.92	6.52	8.28	4.74	23.84
Total	98.68	99.42	100.53	99.90	99.88	98.16	99.92	99.80

	Crescent Graphite	Sweco No. 20	Calcined Coke 3-15	Calcined Coke 3-04	Sweco No. 40	Sweco 60% No. 20 40% No. 40	BB 5
6							
12				.04			
20	.06	.10	.08	.60		0.2	0.14
30	9.36	6.42	23.24	1.44	.02	6.4	0.68
40	11.62	22.34	51.58	3.14	.06	18.0	1.22
50	22.66	52.50	21.42	6.16	1.38	32.0	26.44
70	17.74	18.54	3.36	13.34	10.12	22.2	44.28
100	11.12	1.82	.34	27.24	17.84	10.2	24.62
140	6.36	.08	.06	27.80	16.96	4.2	1.44
200	3.50	.02	.02	12.94	12.86	2.4	0.2
270	1.04			3.0	9.44	1.2	0.03
Pan	6.32		.06	3.32	30.76	3.2	0.68
Total	87.98	102.02	100.16	99.02	99.44	100.	99.78

TABLES

TABLE J5

PROPERTIES OF RAMMED GRAPHITE MOLD MIXTURES

MIXTURE NO.	MOISTURE CONTENT, PER CENT	GREEN PERMEABILITY	GREEN HARDNESS		GREEN COMPRESSION STRENGTH	FIRED TENSILE STRENGTH, PSI	FIRED SCRATCH HARDNESS	
			2 x 2 Specimen	Mold			Test Coupon	Mold
1	7.75	150	77	77	10.3	65	75	75
	8.75	145	76	76	9.4	75	76	90
	8.9	90	78	78	10.3	80	82	82
	10.0	95	78	79	10.5	70	82	94
	9.1	105	77	78	9.7	87	83	95
	9.2	100	77	77	9.8	77	82	96
	8.6	80	79		9.8	75	76	98
	8.8	85	79		10.3	70	78	98
	8	97	77	79	9.5	72.5	80	98
	8.3	97	78	79	8.9	85	80	97
	7.8	83	77		10.1	85	80	97
	7.6	90	78		9.9	90	80	98
	8	93	77		10.1	102.5	82	
	8.1	95	78		10.5	85	83	98
	8.75	95	77		9	75	83	95
	8.5	85	78		9.2	95	83	97
	7.5	82	78		9.2	80	80	98
	8.5	85	79		9.6	90	80	
	8.2	107	77		8.9	95	86	96
	8.4	95	79	80	9.6	95	85	98
	9.0	105	77	78	9.7	80	82	97
	8.2	115	77	80	9.5	65	75	96
	7.4	115	77	80	8.7	67.5	75	
	8	115	78	80	9.4	75	75	97
	7.5	145	77	77	7.2	62.5	75	87
	5.8	160	79	78	9.3	51.5	60	93
	5.5	135	80	80	10.7	75	75	96
	6.5	122	78	78	10.1	60	70	97
	6.75	127	78	80	9.4	50	65	90
3	4.5	170	75	83	5.7	20	20	75
4	6.75	195		80	8.3	45	60	
5	6.2	205		80	8.6	87.5	75	96
6	5.75	155		80	5.5	Broken before test		90
7	9	187		58	4.8	Broken	20	70
8	12	100		48	5.2	20	30	52
9	11.1	155		60	2.9	Broken in firing		0
10	10.2	125		70	4.5	42.5	65	85
11	10.2	90		70	5.9			80
12	9.4	180		70	3.2	Broken	0	20
13	10.3	120		76	6.2	Broken in firing		75
14	10	270		76	4.8	Broken in firing		40
15	9	267		76	4.4	Broken in setup of test		40
16	5.75	270		82	6.7	55	60	93
17	7.6	105	80	80	10.6	46	60	
	6	110	78	80	9.7	50	62	98
	6.3	125	81	80	10.3	47.5	70	98
	6.3	127	79	80	9.7	46	68	97
	7.0	97	79	80	9.9	40	63	97
	6.4	117	79	76	9.5	46	70	96

TABLES

TABLE J5

PROPERTIES OF RAMMED GRAPHITE MOLD MIXTURES

MIXTURE NO.	MOISTURE CONTENT, PER CENT	GREEN PERMEABILITY	GREEN HARDNESS		GREEN COMPRESSION STRENGTH	FIRED TENSILE STRENGTH, PSI	FIRED SCRATCH HARDNESS	
			2 x 2 Specimen	Mold			Test Coupon	Mold
17	7.2	122	78	76	8.0	43	58	95
	7.5	127	79	82	8.6	44	70	
cont.	6.75	150	77	76	7.5	37.5	55	97
	6.75	140	76	80	7.2	47.5	65	
8	152	76	78	6.7	50	70	97	
	7.0	125	77	80	7.1	62.5	70	95
7.4	130	77	80	7.4	62.5	70	95	
	7.3	230	80	7.2	60	65		
7.3	320	80	7.1	50	60			
	6.75	270	77	5.3	50	60		
6.75	290	77	5.3	45	55			
	6.75	280	78	5.8	40	50		
7.1	330	78	5.9	30	55			
	7.0	310	79	6.3				
6.85	330	79	6.9	40	65			
	7.5	300	78	6.5	35	60		
6.85	270	79	7.5	52.5	60			
	6.5	220	79	6.5	60	70		
7.9	210	80	6.5	60	70			
	7.6	270	78	7.3	55	70		
7.5	230	79	6.8	60	70			
	7.75	200	79	7.0	70	75		
7.75	290	80	7.3	65	70			
	7.75	260	79	6.2	52.5	70		
8.0	290	79	5.1	40	55			
	7.9	270	80	6.9	55	65		
7.7	250	79	7.7	65	70			
	8.2	250	79	7.0	60	65		
8.0	240	80	5.7	50	65			
	8.1	230	79	7.3	60	70		
8.0	230	80	7.0	52.5	65			
	8.5	250	78	6.7	Broke	30		
8.8	170	80	7.6	27.5	60			
	8.0	180	79	6.8	35	60		
8.3	180	78	6.9	25	55			
	9.0	240	79	6.3	30	60		
8.5	240	77	4.9					
	7.8	220	79	6.3	30	50		
7.75	250	79	7.1	25	65			
	8.0	260	79	6.9	37.5	70		
8.2	260	78	5.8	22.5	60			
	8.5	240	78	7.1	35	70		
7.4	260	79	6.0					
	7.2	250	79	5.5				
8.6	290	76	4.3					
	7.5	210	79	6.3				
8.3	260	78	5.3					
	7.6	230	79	6.7				
8.0	210	79	6.7					

TABLES

TABLE J5

PROPERTIES OF RAMMED GRAPHITE MOLD MIXTURES

MIXTURE NO.	MOISTURE CONTENT, PER CENT	GREEN PERMEABILITY	GREEN HARDNESS		GREEN COMPRESSION STRENGTH	FIRED TENSILE STRENGTH, PSI	FIRED SCRATCH HARDNESS	
			2 x 2 Specimen	Mold			Test Coupon	Mold
17 cont.	8.0	250	78		5.0			
	8.0	220	77		4.5			
	8.0	230	78		4.6			
	7.2	260	78		4.0			
	7.5	230	78		5.5			
	8.0	270	78		6.1			
	7.75	220	78		6.4			
	7.0	240	79		6.8			
	6.2	95	79		9.7	85	80	
	7.5	105	81		9.3	60	70	
	7.5	95	80		9.4	67.5	75	
	6.9	125	80		9.1	80	77	
	7.5	117	79		8.9	75	75	
	7.8	120	80		8.1	67.5	75	
	7.8	110	80		9.9	77.5	80	
	7.7	120	80		9.7	75	77	
	7.75	113	80		9.8	77.5	80	
	8.0	120	79		8.3	60	70	
	7.9	123	79		8.1	82.5	80	
	7.75	127	80		9.9	90	75	
	7.9	130	80		9.8	70	75	
	7.5	115	78		6.8	65	75	
	7.9	155	80		9.2	70	77	
	7.2	165	80		9.1	50	65	
	7.5	162	80		9.2	55	70	
	7.5	177	79		7.6	70	75	
	8.3	127	79		9.2	77.5	80	
	8.4	152	80		9.1	45	70	
	8.1	145	79		9.8	60	75	
	8.1	190	80		9.6	70	75	
	8.0	245	79		8.9	72.5	75	
	8.2	245	80		8.4	60	70	
	8.4	195	80		9.0	67.5	75	
	7.4	185	80		9.1	50	65	
	7.8	225	79		8.0	62.5	70	
	7.2	200	80		8.5	67	72	
	7.75	215	80		7.7	52.5	70	
	6.25	192	80		8.6	80	75	
	7.0	220	80		8.1	75	77	
	7.2	147	77	80	8.3	77.5	75	92
	6.75	150	79	80	9.3	70	75	
	6.9	120	79	80	9.0	70	75	
	8.6	115	74	80	7.5	64	85	97
18	6.6	105	78	79	6.9	26	35	92
19	7.7	122	75	76	6.1	27.5	40	92
	6.4	132	75	72	6.2	27.5	45	93
20	7.2	117	72	72	5.6	40	55	95
	9.5	105	70	72	4.4	46	70	95

TABLES

TABLE J5

PROPERTIES OF RAMMED GRAPHITE MOLD MIXTURES

MIXTURE NO.	MOISTURE CONTENT, PER CENT	GREEN PERMEABILITY	GREEN HARDNESS		GREEN COMPRESSION STRENGTH	FIRED TENSILE STRENGTH PSI	FIRED SCRATCH HARDNESS	
			2 x 2 Specimen	Mold			Test Coupon	Mold
21	9.3	105	72	72	5.0	40	60	95
	8.1	107	73	72	4.9	35	55	95
	8.5	90	72	78	4.8	52.5	70	96
	9	80	72	72	5.2	61	75	95
	9.4	82	71	72	5.5	50	72	95
	8.8	90	70		4.3	46	70	
	9.2	75	57	78	5.8	137.5	90	95
	9.5	90	72	78	4.8	167.5	90	95
	8	77	74	76	6.7	116.5	85	94
	9	92	73	76	5.6	75	80	
	6.1	440	79	80	8.8	52.5	70	95
	7.2	100	75	73	5.2	35	55	87
22	7.2	100	75	73	5.2	35	55	87
23	8.2	107	72	77	4.3	20	25	94
	8.6	95	74	76	3.8	22.5	30	92
24	8.2	100	73	77	4.4	17.5	20	78
	8.5	110	70	76	2.7	30	50	80
25	9.5	110	73	78	4.1	12.5	10	72
	9	75	74	72	4.9	60	70	91
26	9.5	87	74	76	4.7	21	30	92
	7.75	105	75	80	5.7	20	30	85
27	7.3	87	75	78	5.1	27.5	40	80
	8.2	92	74	78	4.0	20	25	94
28	8.4	73	76	81	6.0	31.0	55	87
	8.2	77	74	81	6.3	32.5	55	87
29	7.5	73	75	78	5.5	32.5	60	93
	9.2	73	74	78	5.2	60	75	94
30	8.75	85	73	78	5.3	52.5	70	94
	8.0	150	80	80	10.6	41.5	65	
30	7.8	200	78	78	10.5	42.5	70	
	7.5	140	77	80	10.8	57.5	80	93
30	7.5	92	78	80	8.7	66.5	80	95
	6.5	88	80	80	9.1	45	70	
30	7.4	62	78	80	9.2	60	80	
	7.8	125	80		10.0	55	65	
30	7.2	130	80		9.8	45	60	
	9.2	97	80		9.0	35	60	
30	10.8	80	78		10.3	60	80	
	9.3	127	80		9.7	40	65	
30	9.25	165	79		9.3	45	75	
	7.2	330	79		6.7			
30	7.3	410	80		7.2			
	7.75	140	78		6.2			
30	9.4	80	80		7.9			
	9.4	75	75	80	7.1	25	55	
31	8.2	75	75	80	8.1	35	75	94
	9.7	60	75	60	8.4	37.5	75	
31	9.75	77	75		8.5			
	9.5	72	76		8.6			

TABLE J5

PROPERTIES OF RAMMED GRAPHITE MOLD MIXTURES

MIXTURE NO.	MOISTURE CONTENT, PER CENT	GREEN PERMEABILITY	GREEN HARDNESS		GREEN COMPRESSION STRENGTH	FIRED TENSILE STRENGTH, PSI	FIRED SCRATCH HARDNESS	
			2 x 2 Specimen	Mold			Test Coupon	Mold
31	9.5	72	76		8.6			
cont.	9.9	75	76		8.1			
32	9.4	68	76	78	8.1	38.5	75	97
33	8.7	100	77	78	9.0	35	80	94
34	9.8	75	75	80	9.0	42.5	80	95
35	9.2	77	75	80	8.4	41.5	78	96
36	8.9	93	75	80	8.6	46	75	95
37	10.0	95	75		9.2	40	75	96
38	10.7	78	74	80	8.7			92
39	14.5	17	70		6.5	297	91	
40	10.0	90	72		5.5	70	77	
	10	93	76		6.8	58.75	65	
41	10.3	46	74		5.8	112.5	85	
42	7.8	107	78		9.9	50	72	
	8.4	97	77		10.4	65	78	
	6.9	147	78		9.2	22.5	40	
43	11.3	38	74		8.5	35	65	
44	8.6	31	79		11.1	150	90	
45	8.1	45	79		10.3	152	90	
46	8.0	95	78		9.0	91	80	
47	12.0	75	75		7.5	135	90	
48	6.5	70	80		8.3	102.5	80	
49	7.5	95	80		8.7	166	87	
	8.75	85	80		9.3	145	87	
	7.5	110	79		9.2	87.5	78	
	8.25	110	79		9.2	102.5	85	
	8.20	100	79		8.8	100	85	
	8.2	110	80		8.9	100	85	
	7.9	125	78		7.2	95	80	
	7.9	120	78		8.6	92.5	82	
	8.25	120	78		8.0	100	82	
	7.9	110	79		8.5	107.5	82	
	8.1	130	80		8.2	112.5	84	
	8.1	130	80		8.7	110	80	
	8.1	93	78		9.4	140	85	
	8.5	93	80		9.6	136.5	85	
	7.0	73	80		10.9	169	87	
	7.8	77	80		10.2	115	80	
	7.5	92	80		9.0	166	87	
	8.75	85	79		9.3	145	87	
	7.5	110	79		9.2			
	8.25	110	79		9.1			
	8.2	100	79		8.7			
	8.2	105	79		8.7			
	7.9	122	78		7.3			
	7.9	115	79		8.0			
	8.25	120	78		8.7			

TABLE J5

PROPERTIES OF RAMMED GRAPHITE MOLD MIXTURES

MIXTURE NO.	MOISTURE CONTENT, PER CENT	GREEN PERMEABILITY	GREEN HARDNESS		GREEN COMPRESSION STRENGTH	FIRED TENSILE STRENGTH, PSI	FIRED SCRATCH HARDNESS	
			2 x 2 Specimen	Mold			Test Coupon	Mold
49	7.9	105	79		8.6			
cont.	8.1	125	79		8.4			
	8.1	125	80		8.8			
50	8.3	410	80		9.4	60	75	
51	11.4	115	75		4.8	72.5	75	
52	7.8	77	75		6.7	80	80	
53	6.2	660	76		5.6	35	60	
54	8.7	142	76		6.1	50	65	
55	7.4	143	76		6.1	42.5	55	
56	9.2	102	75		6.2	40	70	
57	11.5	160	73		6.7	45	65	
58	12.7	85	71		4.9	60	80	
59	12.75	83	70		3.7	95	85	
60	14.5	27	74		7.1	85	87	
61	10.7	270	76		9.5	105	85	
62	7.75	390	78		7.6	35	35	
63	8.2	390	78		8.6	42.5	70	
64	5.4	800	70		4.3			
67	7.6	170	78		7.4			
68	7.8	80	78		6.3	72.5	80	
69	8.6	60	73		4.3	54.0	88	
70	8.0	78	75		4.9	42.0	78	
71	2.36	50	66		3.0			
72	3.34	50	65		2.6	120.0	88	
73	2.10	50	65		2.5	95.0	88	
74		31	69		3.4	218.0	96	
75			70		2.9	200.0	96	
76			66		2.5	94.0	83	
77			76		5.2	77.0	86	
79			75		4.7	56.5	75	
80			76		4.7	81.0	83	
81			79		4.8	61.0	79	
82			77		4.5			
83	11.8	32	73		6.1	33.0	42	
84	12.8	19	75		7.5	25.5	41	
85	13.6	18	78		8.1	33.8	54	
86	12.0	37	74		5.4	57.5	76	
87	13.2	36	71		5.2	46.5	74	
88	11.4	23	75		6.4	47.7	73	
89	9.5	39	69		5.9	49.8	74	
90	8.8	48	72		5.2	34.0	56	
91	8.0	32	70		5.4	36.7	43	
92	15.4	7.5	79		12.3	60	92	
93	15.2	6.5	80		12.7			
94	16.7	7	80		12.2	34	84	
95	13.3	21	75		7.7			
96	12.5	49	65		6.3			
97	12.0	57	65		6.8			
98	10.8	57	64		5.0			

TABLE J5

PROPERTIES OF RAMMED GRAPHITE MOLD MIXTURES

MIXTURE NO.	MOISTURE CONTENT, PER CENT	GREEN PERMEABILITY	GREEN HARDNESS		GREEN COMPRESSION STRENGTH	FIRED TENSILE STRENGTH, PSI	FIRED SCRATCH HARDNESS	
			2 x 2 Specimen	Mold			Test Coupon	Mold
99	11.9	71	65		6.5			
100	11.8	30	67		6.4			
101	11.8	5	81		11.7	49.4	80	
102	14.8	6	81		12.9	55.0	84	
103	14.8	6	80		12.7	82.5	89	
104	14.0	6.5	78		10.2	51.0	88	
105	12.8	7	78		10.2	67.5	85	
106	13.4	6.5	77		10.0	72.0	81	
107	11.8	35	72		5.7	55.0	64	
108	12.2	36	77		6.1	55.0	59	
109	11.0	40	75		5.4	47.5	66	
110	10.6	5	84		15.0		66	
110A	15.6	9	77		11.2	35.0	83	
110B	14.2	13	77		10.5	26.0	74	
111	13.8	6	82		15.9	45.1	77	
112	13.6	6	83		16.0	43.0	80	
113	12.0	40	74		5.3	44.0	71	
114	13.0	41	75		5.9	40.0	70	
115	12.0	46	72		4.9	31.2	40	
116	13.6	40	69		4.8	77.5	85	
117	12.6	48	68		4.4	62.5	88	
118	11.4	59	70		5.0	48.5	74	
119	9.2	51	69		3.4	32.5	53	
120	10.3	52	68		3.6			
121	9.8	58	70		3.4			
122	17.6	5.5	80		13.7		88	
123	19.6	5.5	80		12.6	82.5	92	
124	21.8	5	80		12.5		93	
125	16.8	7	75		9.0	41.0	86	
126	13.8	6	75		9.8	69.0	89	
127	15.6	6.5	75		9.7	122.0	97	
128	13.4	31	70		4.9	58.5	77	
129	10.2	50	68		3.0	42.0	55	
130	10.8	54	73		4.4	39.5	52	
131	9.2	47	72		5.4	48.5	72	
132	9.6	48	70		5.0	56.0	80	

TABLE J6

SHRINKAGE OF RAMMED GRAPHITE MOLDS

MIX NO.	SHRINKAGE, IN/FT	MIX NO.	SHRINKAGE, IN/IN	MIX NO.	SHRINKAGE, IN/FT
2	0.128	17	0.0110 0.01185 0.0122 0.0120 0.0145 0.0145 0.00805 0.0145 0.0145 0.0145 0.0126	19	0.207 0.173 0.190
4	0.198			Ave.	
5	0.198			20	0.173 0.132 0.152
16	0.188			Ave.	
17	0.125 0.125 0.138 0.136 0.136 0.133 0.140 0.133	Ave.		21	0.138 0.173 0.160 0.159 0.0834 0.146 0.0612 0.131
		30	0.0136 0.0126 0.0131		
Ave.		Ave.			
22	0.188	40	0.00745 0.01435 0.00109	Ave.	
30	0.141 0.141 0.141	41	0.00825	23	0.114
Ave.		43	0.00885	24	0.132 0.151 0.141
32	0.151	44	0.0151	Ave.	
33	0.155	46	0.01385	25	0.104 0.0345 0.0694
6	0.139	47	0.0153	Ave.	
31	0.138 0.104 0.0926 0.0909 0.0792 0.101	48	0.0142	26	0.076 0.045 0.104 0.075
Ave.		49	0.01535 0.01450 0.01450 0.01450 0.01450 0.01453	Ave.	
34	0.0755	Ave.		27	0.0755 0.0755 0.0755
35	0.104	51	0.01025	Ave.	
36	0.1035	54	0.0152	28	0.1385 0.136 0.151 0.149
37	0.0755	55	0.0141	Ave.	
38	0.138	56	0.0103	29	0.166
		57	0.0107		
		58	0.0149		

TABLE J7

EXPERIMENTAL SHELL GRAPHITE MOLD MIXTURES

Mix No.	Phenol Formal- dehyde %	Pitch %	Graphite Type	%	Solvent	Mull Time, Minutes
1	12		BB 5	88		4
2	12		Swaco #40	88		5
3	20		BB 5	80		5
4	12	8	BB 5	80		5
5	6	8	BB 5	86	1/2 Pt.	4*
6	12	8	BB 5	80	1/2 Pt.	4*
7	10 (Borden)	10	Carbon Sand	80		4
8	12 (7504)	10	Calcined Coke	78		4
9	10 (7504)	10	Calcined Coke	78		4

* Mullied dry 4 minutes and then mullied until alcohol solvent evaporated.

TABLE J8

EXPERIMENTAL INVESTMENT MOLD MIXTURES

Mix No.	Graphite	Darex 561	Darex 32	Commercial White Silica	C-3 Cement	Pitch	Water	Comments
	<u>#50 Mesh</u>	<u>Grade 39</u>						
1	5 Parts	4 Parts	2½ Parts					Medium shrinkage 20 hr. set time
2	7 Parts	2 Parts	2 Parts					Medium shrinkage 8 hr. set time
3	7 Parts		2 Parts	2 Parts				High shrinkage strong, 10 hr. set time.
4	7 Parts	2 Parts					1 Part	10 hr. set time hard strength no shrinkage
5	7 Parts			2 Parts			2 Parts	16 hr. set time, hard - no shrinkage
6	70 GM	20 GM			8 GM	10 GM	25 GM	Mold cavity pitted
7	70 GM	20 GM	20 GM		8 GM	10 GM		Mold cavity badly pitted - poor condition
8	70 GM			15 GM	8 GM	10 GM	30 GM	Scabbing, partial collapse of mold walls
9	70 GM	10 GM		6 GM (starch)	8 GM	10 GM	50 GM	Very porous, cavity collapsed.
10	350 GM	100 GM			40 GM	50 GM	125 GM	Air entrapped at cavity surfaces thick slurry.
11 (H-1) (First Coat)	30 GM	40 GM	30 GM		16 GM	20 GM	90 GM	First coat buckled and pulled away from second coat, air entrapped in second coat.
11A (Second Coat)	350 GM	50 GM			40 GM	50 GM	140 GM	
12	350 GM	50 GM			40 GM	50 GM	120 GM	Too stiff to pour well
13	400 GM	150 GM	40 GM			50 GM	270 GM	Holds good. Some air entrapment
14A (F-1) (First coat)	50 GM	30 GM	12 GM		10 GM	12 GM	65 GM	
14B (F-1)	100 CC Tetraethyl orthosilicate, 96 CC denatured ethyl alcohol 24 CC 5% hydrochloric acid. (plus 12 CC of water to control setting time) Added to: 200 GM #20 Grog, 210 GM #120 mesh Nevada sand, 120 GM - 200 mesh silica flour, 1 GM -magnesium oxide.							
15 (H-1)	Kerr crystallite and water							
16 (G-1)	200 GM	150 GM	50 GM		40 GM	50 GM	270 GM	Best mix

TABLES

TABLE J9
GRAIN SIZE DISTRIBUTION OF RECYCLED MOLD MATERIALS

Sample No.	Per cent Retained on Screens											
	1	2	3	4	5	6	7	8	9	10	11	12
1						0.4	0.4	0.2	0.4	0.2	0.25	0.2
2					0.8	2.4	2.2	1.8	2.8	3.2	3.24	1.2
3			0.2	1.0								0.4
4	.184	0.2	1.4	2.4	2.6	3.2	3.2	3.2	5.4	6.4	7.18	3.2
5	1.500	2.2	3.4	5.2	5.6	6.8	5.8	7.6	8.6	11.0	13.34	9.6
6	7.636	9.2	10.2	13.8	14.4	15.6	15.4	18.6	21.2	22.8	25.86	22.4
7	16.464	8.0	17.6	20.6	20.6	20.6	21.6	24.2	24.4	25.2	24.12	23.0
8	16.726	8.4	17.8	19.0	18.0	17.4	16.6	16.6	15.6	13.0	11.20	13.4
9	16.970	1.2	11.6	10.6	10.1	9.4	9.2	7.2	6.6	5.4	3.36	6.2
10	6.440	8.0	8.8	7.0	7.2	6.4	6.6	5.4	4.2	3.2	2.40	4.6
11	.160	0.2	0.4	2	—	0.4	0.2	0.2	—	—	.04	1.0
12	30.320	17.8	16.6	20.0	5.6	4.0	6.8	0.2	3.2	8.4	1.54	2.8
13	92.40	100	88.2	100	85.0	66.4	90.2	85.4	92.6	96.8	92.62	87.2
14	.02718	.02318	3/64in-SL	1/16in	3/32in	1/8in	3/16 in	3/16 in	3/16 in	3/16 in	1/2 in	100
15	14.000	12.100	12.100	12.100	12.100	12.100	12.100	8.050	8.050	6.070	6.070	6.070
16												Calined Surco #20
17												Cone Graphite
18												Fracture

UNLESS OTHERWISE NOTED IS FINED GRAPHITE.

TABLE J10

FEEDING DISTANCES IN UN-TAPERED MACHINED GRAPHITE MOLDS

HEAT NO.	PLATE			UNDER RISER SHRINK	PLATE SHRINK	EDGE SOUND	SOUND BEYOND RISER	TOTAL SOUND	Y =	
	THICK-NESS	DIA-METER	RISER DIA.						Total Thickness	Sound
P-7	1 in.	6 in.	1 in.	No	Obs	Yes	5/8 "	7/16	1-1/16	1.0625
	7/8			No	Obs	Yes	0	13/16	13/16	0.929
	3/4			No	Obs	Yes	1/4	5/8	7/8	1.16
	5/8			No	Obs	Yes	3/8	9/16	15/16	1.50
P-9	1 in.	6 in.	1.5 in.	Yes	Yes	Yes	5/8	3/8	1 in.	1.0
	7/8			Yes	Yes	Yes	9/16	5/8	1-3/16	1.36
	3/4			No	Yes	Yes	5/8	5/8	1-1/4	1.67
	5/8			No	Yes	Yes	3/8	7/16	13/16	1.30
P-10	1 in.	6 in.	2 in.	No	Yes	Yes	3/4	1/4	1 in.	1.0
	7/8			No	Yes	Yes	11/16	13/16	1-1/2	1.71
	3/4			No	Yes	Yes	9/16	11/16	1-1/4	1.67
	5/8			No	Yes	Yes	5/16	9/16	7/8	1.40
P-11	1 in.	6 in.	2.5 in.	No	Yes	Yes	1/2	11/16	1-3/16	2.37
	7/8			No	Yes	Yes	9/16	1/2	1-1/16	1.0625
	3/4			No	Yes	Yes	1/4	1/2	3/4	0.75
	5/8			No	Yes	Yes	5/16	5/8	15/16	1.50
P-12	1 in.	5 in.	None	No	Yes	Yes	1/4	5/16	9/16	1.12
	7/8			No	Yes	Yes	1/4	5/16	9/16	1.12
	3/4			No	Yes	Yes	1/4	5/16	9/16	1.12
	5/8			No	Yes	Yes	1/4	5/16	9/16	1.12
P-13	1 in.	5 in.	None	No	Yes	Yes	1/4	5/16	9/16	1.12
	7/8			No	Yes	Yes	1/4	5/16	9/16	1.12
	3/4			No	Yes	Yes	1/4	5/16	9/16	1.12
	5/8			No	Yes	Yes	1/4	5/16	9/16	1.12
P-14	1 in.	5 in.	1 in.	Yes	Yes	Yes	1/2	1/2	1	1.0
	7/8			Yes	Yes	Yes	1	5/16	1-5/16	1.5
	3/4			Yes	Yes	Yes	1/2	1/2	1	1.33
	5/8			Yes	Yes	Yes	9/16	1/2	1-1/16	1.70
P-15	1 in.	5 in.	1.5	Yes	Yes	Yes	7/16	3/8	13/16	1.60
	7/8			No	Yes	Yes	3/8	7/16	13/16	2.15
	3/4			No	Yes	Yes	3/8	7/16	13/16	2.15
	5/8			No	Yes	Yes	3/8	7/16	13/16	2.15
P-16	1 in.	5 in.	2 in.	Yes	Yes	Yes	11/16	7/8	1-9/16	1.56
	7/8			Yes	Yes	Yes	5/8	5/16	15/16	1.06
	3/4			Yes	Yes	Yes	9/16	3/8	15/16	1.25
	5/8			No	Yes	Yes	1/2	5/8	1-1/8	1.8
P-17	1 in.	5 in.	2.5 in.	No	Yes	Yes	1/4	5/16	9/16	1.12
	7/8			No	Yes	Yes	1/8	3/4	7/8	2.30
	3/4			No	Yes	Yes	1/2	5/8	1-1/8	1.125
	5/8			No	Yes	Yes	9/16	1/2	1-1/16	1.2
P-18	1 in.	5 in.	2 in.	No	Yes	Yes	1/2	11/16	1-3/16	1.48
	7/8			No	Yes	Yes	1/2	5/8	1-1/8	1.80
	3/4			No	Yes	Yes	3/8	5/16	11/16	1.37
	5/8			No	Yes	Yes	3/16	3/16	3/8	1.00
P-19	1 in.	5 in.	2.5 in.	No	No	No			1-1/4	1.25
	7/8			No	No	No			1-1/4	1.43
	3/4			No	No	No			1-1/4	1.66
	5/8			No	Yes	Yes	1/2	5/8	1-1/8	1.8
P-20	1 in.	5 in.	2.5 in.	No	Yes	Yes	11/16	7/16	1-1/8	2.25
	7/8			No	Yes	Yes	1/4	1/2	3/4	2.00

TABLE J11

FEEDING DISTANCES IN TAPERED MACHINED GRAPHITE MOLDS

HEAT NO.	PLATE			TAPER	UNDER RISER SHRINK	PLATE SHRINK	EDGE SOUND	SOUND BEYOND RISER	TOTAL SOUND	T s	
	THICK-NESS	DIA-METER	RISER DIA.							Total	Sound Thickness
P-74	1/8 in.	6 in.	1 in.	2	no	yes	1/4	1/4	1/2	4.00	
	1/4	6	1	2	no	yes	1/4	1/2	3/4	3.00	
	3/8	6	1	2	no	yes	3/4	1/2	1-1/4	3.33	
	1/2	6	1.5	2	no	yes	3/8	5/8	1	2.00	
	5/8	6	1.5	2	no	yes	5/8	3/4	1-3/8	2.20	
	3/4	6	2	2	no	yes	3/4	1/2	1-1/4	1.670	
	7/8	6	2	2	no	yes	3/4	1/2	1-1/4	1.440	
P-75	1	6	2.5	2	no	yes	3/4	1/4	1-1/4	1.250	
	1/8 in.	6 in.	1 in.	2	no	yes	1/4	1/4	1/2	4.00	
	1/4	6	1	2	no	yes	1/4	1/2	3/4	3.00	
	3/8	6	1	2	no	yes	1/2	1/4	3/4	2.00	
	1/2	6	1.5	2	no	yes	1/2	1/4	3/4	1.50	
	5/8	6	1.5	2	no	yes	1/2	1/2	1	1.60	
	3/4	6	2	2	no	yes	3/8	1/4	5/8	.834	
P-76	7/8	6	2	2	no	yes	5/8	1	1-5/8	1.86	
	1	6	2.5	2	no	yes	3/4	1/2	1-1/4	1.250	
	1/8 in.	6 in.	1 in.	2	no	yes	1/4	1/4	1/2	4.00	
	1/4	6	1	2	no	yes	1/4	1/4	1/2	2.00	
	3/8	6	1	2	no	yes	1/2	3/8	7/8	2.33	
	1/2	6	1.5	2	no	yes	1/2	1/4	3/4	1.50	
	5/8	6	1.5	2	no	yes	1/2	1/2	1	1.60	
P-77	3/4	6	2	2	no	yes	1/2	0	1/2	.666	
	7/8	6	2	2	no	yes	3/4	1/4	1	1.14	
	1	6	2.5	2	no	yes	1/2	1/4	3/4	.75	
	1/8 in.	6 in.	1 in.	3	no	yes	1/8	3/8	1/2	4.00	
	1/4	6	1	3	no	yes	1/4	1/4	1/2	2.00	
	3/8	6	1	3	no	yes	1/4	1/4	1/2	1.33	
	1/2	6	1.5	3	no	yes	1/2	1/4	3/4	1.50	
P-78	5/8	6	1.5	3	no	yes	1/2	1/2	1	1.60	
	3/4	6	2	3	no	yes	3/8	0	3/8	.50	
	7/8	6	2	3	no	yes	3/4	1/2	1-1/4	1.43	
	1	6	2.5	3	no	yes	3/8	1/2	7/8	.87	
	1/8 in.	6 in.	1 in.	3	no	yes	1/8	1/4	3/8	3.00	
	1/4	6	1	3	no	yes	1/4	3/4	1	4.00	
	3/8	6	1	3	no	yes	1/4	1/4	1/2	1.33	
P-79	1/2	6	1.5	3	no	yes	3/8	1/2	3/8	1.75	
	5/8	6	1.5	3	no	yes	1/2	1/4	3/4	1.20	
	3/4	6	2	3	no	yes	1/2	3/4	1-1/4	1.67	
	7/8	6	2	3	no	yes	5/8	1	1-5/8	1.86	
	1	6	2.5	3	no	yes	5/8	1/2	1-1/8	1.125	
	1/8 in.	6 in.	1 in.	3	no	yes	1/8	1/4	3/8	3.00	
	1/4	6	1	3	no	yes	1/4	1/4	1/2	2.00	
P-80	3/8	6	1	3	no	yes	1/4	1/4	1/2	1.33	
	1/2	6	1.5	3	no	yes	0	0	0	0.00	
	5/8	6	1.5	3	no	yes	1/2	1/2	1	1.60	
	3/4	6	2	3	no	yes	5/8	1/2	1-1/8	1.50	
	7/8	6	2	3	no	yes	7/8	1/2	1-1/4	1.43	
	1	6	2.5	3	no	yes	5/8	3/4	1-3/8	1.37	
	1/8 in.	6 in.	1 in.	4	no	yes	1/4	1/4	1/4	2.00	
P-81	1/4	6	1	4	no	yes	1/2	3/8	7/8	3.50	
	3/8	6	1	4	no	yes	1/2	1/4	3/4	2.00	
	1/2	6	1.5	4	no	yes	3/4	0	3/4	1.50	
	5/8	6	1.5	4	no	yes	5/8	1/4	7/8	1.46	
	3/4	6	2	4	no	yes					
	7/8	6	2	4	no	yes	1/2	1-1/2	2	2.29	
	1	6	2.5	4	no	yes	3/4	3/4	1-1/2	1.50	
P-82	1/8 in.	6 in.	1 in.	4	no	yes	1/4	1/4	1/4	3.00	
	1/4	6	1	4	no	yes	1/4	1/4	1/4	2.00	
	3/8	6	1	4	no	yes	1/4	1/4	1/4	1.00	
	1/2	6	1.5	4	no	yes	1/4	1/4	1/4	1.00	
	5/8	6	1.5	4	no	yes	1/4	1/4	1/4	1.00	
	3/4	6	2	4	no	yes	1/4	1/4	1/4	1.00	
	7/8	6	2	4	no	yes	1/4	1/4	1/4	1.00	
P-83	1	6	2.5	4	no	yes	1/4	1/4	1/4	1.00	
	1/8 in.	6 in.	1 in.	4	no	yes	1/4	1/4	1/4	3.00	
	1/4	6	1	4	no	yes	1/4	1/4	1/4	2.00	
	3/8	6	1	4	no	yes	1/4	1/4	1/4	1.00	
	1/2	6	1.5	4	no	yes	1/4	1/4	1/4	1.00	
	5/8	6	1.5	4	no	yes	1/4	1/4	1/4	1.00	
	3/4	6	2	4	no	yes	1/4	1/4	1/4	1.00	
P-84	7/8	6	2	4	no	yes	1/4	1/4	1/4	1.00	
	1	6	2.5	4	no	yes	1/4	1/4	1/4	1.00	
	1/8 in.	6 in.	1 in.	4	no	yes	1/4	1/4	1/4	3.00	
	1/4	6	1	4	no	yes	1/4	1/4	1/4	2.00	
	3/8	6	1	4	no	yes	1/4	1/4	1/4	1.00	
	1/2	6	1.5	4	no	yes	1/4	1/4	1/4	1.00	
	5/8	6	1.5	4	no	yes	1/4	1/4	1/4	1.00	

TABLE J11 (Continued)

HEAT NO.	PLATE THICKNESS	PLATE DIA-METER	RISER DIA.	TAPER	UNDER RISER SHRINK	PLATE SHRINK	EDGE SOUND	SOUND BEYOND RISER	TOTAL SOUND	T = Total Sound Thickness
P-88	1/8 in.	6 in.	1 in.	4	no	yes	1/8	3/8	1/2	4.00
	1/4	6	1	4	no	yes	1/4	1/4	1/2	2.00
	3/8	6	1	4	no	yes	1/4	3/8	7/8	2.33
	1/2	6	1.5	4	no	yes	1/2	1/4	3/4	1.25
	5/8	6	1.5	4	no	yes	1/2	0	1/2	1.25
	3/4	6	2	4	no	yes	1/2	3/4	1-1/4	1.67
	7/8	6	2	4	no	yes	5/8	1/2	1-1/8	1.28
	1	6	2.5	4	no	yes	3/4	1/2	1-1/4	1.25
P-95	1/8 in.	6 in.	1 in.	5	yes	yes	1/8	3/4	7/8	7.00
	1/4	6	1	5	yes	yes	1/4	1/4	1/2	2.00
	3/8	6	1.5	5	no	yes	1/2	0	1/2	1.33
	1/2	6	1.5	5	no	yes	1/2	1/2	1	2.00
	5/8	6	1.5	5	no	yes	3/8	7/8	1-1/4	2.00
	3/4	6	2	5	no	yes	1/2	3/4	1-1/4	1.67
	7/8	6	2	5	no	yes	5/8	1/4	1-1/8	1.28
	1	6	2.5	5	no	yes	3/4	3/4	1-1/2	1.50
P-101	1/8 in.	6 in.	1 in.	5	yes	yes	1/8	3/4	7/8	7.00
	1/4	6	1	5	yes	yes	1/4	1/4	1/2	2.00
	3/8	6	1.5	5	no	yes	1/4	1/4	1/2	1.33
	1/2	6	1.5	5	no	yes	1/2	1/4	3/4	1.50
	5/8	6	1.5	5	no	yes	3/4	1/2	1-1/4	2.00
	3/4	6	2	5	no	yes	3/8	3/4	1-1/8	1.50
	7/8	6	2	5	no	yes	3/4	1	1-3/4	2.00
	1	6	2.5	5	no	yes	3/4	1/4	1-1/4	1.25
P-105	1/8 in.	6 in.	1 in.	5	yes	yes		1/2	1/2	4.00
	1/4	6	1	5	yes	yes		1/4	1/4	1.00
	3/8	6	1.5	5	no	yes	1/2	0	1/2	1.33
	1/2	6	1.5	5	no	yes	1/4	1/4	1/2	1.00
	5/8	6	1.5	5	no	yes	1/2	1/2	1	1.60
	3/4	6	2	5	no	yes	3/4	3/4	1-1/2	2.00
	7/8	6	2	5	no	yes	3/4	3/4	1-1/2	1.71
	1	6	2	5	no	yes	3/4	3/4	1-1/2	1.25
P-117	1/8 in.	6 in.	1 in.	6	no	yes	1/4	5/8	7/8	7.00
	1/2	6	1	6	no	yes	1/4	1/2	3/4	3.00
	3/8	6	1.5	6	no	yes	1/2	1/4	3/4	2.00
	1/2	6	1.5	6	no	yes	5/8	1/4	7/8	1.75
	5/8	6	1.5	6	no	yes	5/8	1/2	1-1/8	1.80
	3/4	6	2	6	no	yes	5/8	3/4	1-3/8	1.83
	7/8	6	2	6	no	yes	3/4	3/4	1-1/2	1.72
	1	6	2.5	6	no	yes	3/4	1/4	1	1.00
P-120	1/8 in.	6 in.	1 in.	6	no	yes	1/8	1/2	5/8	5.00
	1/4	6	1	6	no	yes	1/4	1/2	3/4	3.00
	3/8	6	1.5	6	no	yes	1/2	0	1/2	1.33
	1/2	6	1.5	6	no	yes	1/2	1/4	3/4	1.50
	5/8	6	1.5	6	no	yes	1/2	yes	1/2	.80
	3/4	6	2	6	no	yes	1/2	1/4	3/4	1.00
	7/8	6	2	6	no	yes	1/2	1	1-1/2	1.73
	1	6	2.5	6	no	yes		3/4		
P-128	1/8 in.	6 in.	1 in.	6	no	yes	1/4	1/2	3/4	6.00
	1/4	6	1	6	no	yes	1/4	1/4	1/2	2.00
	3/8	6	1.5	6	no	yes	3/8	1/4	5/8	1.66
	1/2	6	1.5	6	no	yes	1/2	1/4	3/4	1.50
	5/8	6	1.5	6	no	yes		1/2	1/2	.80
	3/4	6	2	6	no	yes	1/2	1/4	3/4	1.00
	7/8	6	2	6	no	yes	3/4	1/4	1-1/4	1.67
	1	6	2.5	6	no	yes		0		

TABLE J11 (Continued)

Heat No.	PLATE					Under		Sound			Total Sound	Total Sound Thickness.
	Thick-ness	Dia-meter	Riser Dia.	Riser Type	Riser Shrink	Plate Shrink	Edge Sound	Behind Riser				
P143	1/8 in.	6 in.	1 in.	7	yes	yes	1/4	1/4	1/2	4.00		
	1/4	6	1.5	7	no	yes	1/4	1/4	1/2	2.00		
	3/8	6	1.5	7	no	yes	1/4	0	1/4	.66		
	1/2	6	2	7	no	yes	1/2	5/8	1 1/8	2.25		
	5/8	6	2	7	no	gas	-	-	-	-		
	3/4	6	2.5	7	no	gas	-	-	-	-		
	7/8	6	2.5	7	no	yes	3/4	3/4	1 1/2	1.80		
1	6	3	7	no	no	-	-	1 1/2	1.50			
P153	1/8 in.	6 in.	1 in.	7	yes	yes	3/16	3/8	5/16	4.51		
	1/4	6	1.5	7	no	yes	1/4	1/4	1/2	2.00		
	3/8	6	1.5	7	no	yes	1/2	1/2	1	2.67		
	1/2	6	2	7	no	yes	1/4	1/8	3/8	.75		
	5/8	6	2	7	no	gas	-	-	-	-		
	3/4	6	2.25	7	no	gas	5/8	1/2	1 1/8	1.50		
	7/8	6	2.5	7	no	gas	-	-	-	-		
1	6	3	7	no	gas	-	-	-	-			
P154	1/8 in.	6 in.	1 in.	7	no	yes	1/4	1/4	1/2	4.00		
	1/4	6	1.25	7	no	yes	gas	1/4	1/4	1.00		
	3/8	6	2	7	yes	yes	3/4	3/8	1 1/8	3.00		
	1/2	6	2	7	no	yes	3/4	1/2	1 1/4	2.50		
	5/8	6	2	7	no	yes	3/4	5/8	1 3/8	2.20		
	3/4	6	2.25	7	no	yes	5/8	5/8	1 1/4	1.67		
	7/8	6	2.25	7	no	gas	-	-	-	-		
1	6	3	7	no	no	-	-	1 1/2	1.50			
P161	1/8 in.	6 in.	1 in.	8	no	yes	1/8	3/8	1/2	4.00		
	1/4	6	1.5	8	no	yes	1/4	1/4	1/2	2.00		
	3/8	6	1.5	8	no	yes	1/4	1/4	1/2	1.33		
	1/2	6	2	8	no	yes	5/8	3/4	1 3/8	2.75		
	5/8	6	2	8	no	yes	1 1/8	3/8	1 1/2	2.40		
	3/4	6	2.875	8	no	yes	3/4	1/4	1	1.33		
	7/8	6	2.5	8	no	yes	5/8	3/4	1 3/8	1.65		
1	6	3	8	no	no	-	-	1 1/2	1.50			
P162	1/8 in.	6 in.	1 in.	8	no	yes	3/16	1/4	7/16	3.50		
	1/4	6	1.5	8	no	yes	1/4	3/16	7/16	1.75		
	3/8	6	1.5	8	no	yes	1/4	1/4	1/2	1.33		
	1/2	6	2	8	no	yes	5/8	3/4	1 3/8	2.75		
	5/8	6	2	8	no	yes	3/8	1/2	7/8	1.40		
	3/4	6	2.5	8	no	no	-	-	1 3/4	2.33		
	7/8	6	2.5	8	no	no	-	-	1 3/4	2.10		
1	6	3	8	no	no	-	-	1 1/2	1.50			
P163	1/8 in.	6 in.	1.5 in.	8	no	yes	gas	1/4	1/4	4.00		
	1/4	6	1.5	8	no	yes	gas	1/4	1/4	1.00		
	3/8	6	1.5	8	no	yes	3/4	1/2	1 1/4	3.33		
	1/2	6	2	8	no	yes	3/4	3/16	15/16	1.47		
	5/8	6	2	8	no	yes	1/2	3/4	1 1/4	2.00		
	3/4	6	2.875	8	no	yes	5/8	3/8	1	1.33		
	7/8	6	2.5	8	no	yes	3/4	1/4	1 1/4	1.50		
1	6	3	8	no	no	-	-	1 1/2	1.50			
P177	1/8 in.	6 in.	1.5 in.	9	yes	yes	3/16	1/4	7/16	5.50		
	1/4	6	1.5	9	no	yes	1/4	1/4	1/2	4.00		
	3/8	6	1.5	9	no	yes	3/8	1	1 3/8	3.67		
	1/2	6	2	9	no	yes	7/8	1/4	1 1/8	2.75		
	5/8	6	2	9	no	yes	1 1/8	1/8	1 1/2	2.40		
	3/4	6	2.25	9	no	yes	5/8	3/4	1 3/8	1.83		
	7/8	6	2.5	9	no	yes	1	3/4	1 3/4	2.10		
1	6	3	9	no	no	-	-	1 1/2	1.50			

TABLE J11 (Continued)

Heat No.	Thick- ness	PLATE				Under Risor Shrink	Plate Shrink	Edge Shrink	Sound Beyond Risor	Total Sound	Total Sound Thickness
		Dia- meter	Risor Dia.	Taper							
P178	1/8 in.	6 in.	1.5 in.	9	yes	yes	1/4	1/2	3/4	6.00	
	1/4	6	1.5	9	no	yes	1/4	1/4	1/2	2.00	
	3/8	6	1.5	9	no	yes	1/4	1/4	1/2	1.33	
	1/2	6	2	9	no	yes	5/8	3/8	1	2.00	
	5/8	6	2	9	no	yes	7/8	3/8	1 1/4	2.00	
	3/4	6	2.25	9	no	yes	7/8	1/2	1 3/8	1.83	
	7/8	6	2.5	9	no	yes	3/4	3/4	1 1/2	1.80	
	1	6	3	9	no	no	-	-	1 1/2	1.50	
P179	1/8 in.	6 in.	1 in.	9	yes	yes	1/8	1/4	3/8	3.00	
	1/4	6	1.5	9	no	yes	1/4	3/8	5/8	2.50	
	3/8	6	1.5	9	no	yes	1/2	1/4	3/4	2.00	
	1/2	6	2	9	no	yes	3/4	1/2	1 1/4	2.50	
	5/8	6	2	9	no	yes	3/4	3/4	1 1/2	2.40	
	3/4	6	2.25	9	no	yes	1 1/4	1/2	1 3/4	2.33	
	7/8	6	2.5	9	no	no	-	-	1 3/4	2.10	
	1	6	3	9	no	no	-	-	1 1/2	1.50	
P208	1/8 in.	6 in.	1 in.	10	yes	yes	3/8	1/4	5/8	5.00	
	1/4	6	1.5	10	no	yes	3/8	1/4	5/8	2.50	
	3/8	6	1.5	10	yes	yes	3/8	1/2	7/8	2.43	
	1/2	6	2	10	no	yes	3/4	3/4	1 1/2	3.00	
	5/8	6	2	10	no	yes	1	5/8	1 5/8	2.60	
	3/4	6	2.5	10	no	yes	1	1/4	1 1/4	1.66	
	7/8	6	2.5	10	no	yes	3/4	1/2	1 1/4	1.50	
	1	6	3	10	no	no	-	-	1 1/2	1.50	
P212	1/8 in.	6 in.	1 in.	10	yes	yes	3/16	3/8	9/16	4.50	
	1/4	6	1.5	10	no	yes	1/2	1/4	3/4	3.00	
	3/8	6	1.5	10	no	yes	3/8	3/8	5/4	2.22	
	1/2	6	1.5	10	no	yes	1	1/2	1 1/2	3.00	
	5/8	6	2	10	no	yes	5/8	5/8	1 1/4	2.00	
	3/4	6	2.5	10	no	no	-	-	1 3/4	2.33	
	7/8	6	2.5	10	no	yes	1/2	5/8	1 1/8	1.34	
	1	6	3	10	no	no	-	-	1 1/2	1.50	
P215	1/8 in.	6 in.	1.5 in.	10	yes	yes	gas	1/8	1/8	1.00	
	1/4	6	1.5	10	no	yes	3/8	3/8	3/4	3.00	
	3/8	6	1.5	10	no	yes	1/2	3/8	7/8	2.33	
	1/2	6	2	10	no	yes	1	3/4	1 3/4	3.50	
	5/8	6	2	10	no	yes	7/8	1/2	1 3/8	2.70	
	3/4	6	2.5	10	no	yes	5/8	1	1 5/8	2.17	
	7/8	6	2.5	10	no	yes	3/4	1/2	1 1/4	1.43	
	1	6	3	10	no	no	-	-	1 1/2	1.50	
P269	1/8 in.	6 in.	1 in.	11	yes	yes	1/4	1/2	3/4	6.00	
	1/4	6	1.5	11	no	yes	3/8	1/2	7/8	2.16	
	3/8	6	1.5	11	no	yes	1/2	1/2	1	2.66	
	1/2	6	2	11	no	yes	1	3/8	1 3/8	2.75	
	5/8	6	2.25	11	no	no	-	-	1 7/8	3.00	
	3/4	6	2.5	11	no	yes	1 1/8	3/8	1 1/2	2.00	
	7/8	6	2.5	11	no	yes	1	1/2	1 1/2	1.71	
	1	6	3	11	no	gas	-	-	-	-	
P273	1/8 in.	6 in.	1 in.	11	yes	yes	1/16	1/4	5/16	2.50	
	1/4	6	1.5	11	no	yes	3/8	1/4	5/8	2.50	
	3/8	6	1.5	11	no	yes	1	1/2	1 1/2	4.00	
	1/2	6	2	11	no	yes	1 1/8	1/4	1 3/8	2.74	
	5/8	6	2.25	11	no	yes	3/4	1	1 3/4	2.60	
	3/4	6	2.5	11	no	gas	-	-	-	-	
	7/8	6	2.25	11	yes	gas	1/4	1/4	7/8	.857	
	1	6	3	11	no	no	-	-	1 1/2	1.50	
P311	1/8 in.	6 in.	1.5 in.	12	yes	gas	-	-	-	-	
	1/4	6	1.375	12	no	yes	1/4	7/8	1 1/8	4.50	
	3/8	6	2	12	no	yes	1 1/4	5/8	1 7/8	5.00	
	1/2	6	2.5	12	no	yes	1 3/16	3/4	1 9/16	3.13	
	5/8	6	2.5	12	no	no	-	-	1 3/4	4.40	
	3/4	6	2.5	12	no	no	-	-	1 3/4	3.64	
	7/8	6	2.5	12	no	no	-	-	1 3/4	3.15	
	1	6	3	12	no	no	-	-	1 1/2	1.50	

TABLE J12

FEEDING DISTANCES IN TAPERED RAMMED GRAPHITE MOLDS

HEAT NO.	THICK- NESS	PLATE DIA- METER	RISER DIA.	TAPER	UNDER RISER SHRINK	PLATE SHRINK	EDGE SOUND	SOUND BEYOND RISER	TOTAL SOUND	TOTAL Sound Thickness
P-90	1/8 in.	6 in.	1 in.	0	yes	yes	1/8"	1/8	1/4	2.00
	1/4	6	1	0	yes	yes	1/4	1/8	3/8	1.50
	3/8	6	1	0	yes	yes	1/4	1/4	1/2	1.33
	1/2	6	1.5	0	yes	yes	1/4	0	1/4	.50
	5/8	6	1.5	0	yes	yes	1/2	3/8	7/8	1.40
	3/4	6	2	0	yes	yes	1/2	1	1-1/2	2.00
	7/8	6	2	0	no	yes	3/8	1/2	5/8	.71
	1	6	2.5	0	yes	no	--	4-3/4		
P-92	1/8 in.	6 in.	1 in.	0	yes	yes	1/16	1/8	3/16	1.50
	1/4	6	1	0	yes	yes	1/4	1/4	1/2	2.00
	3/8	6	1	0	yes	yes	1/4	1/4	1/2	1.33
	1/2	6	1.5	0	yes	yes	1/4	1/16	5/16	.624
	5/8	6	1.5	0	yes	yes	3/8	5/16	11/16	1.09
	3/4	6	2	0	no	yes	3/8	1/2	7/8	1.166
	7/8	6	2	0	no	yes	1/2	3/4	1-1/4	1.43
	1	6	2.5	0	yes			gas		
P-93	1/8 in.	6 in.	1 in.	0	yes	yes	1/8	1/8	1/4	2.00
	1/4	6	1	0	yes	yes	1/4	1/4	1/2	2.00
	3/8	6	1	0	yes	yes	1/4	1/4	1/2	1.33
	1/2	6	1.5	0	yes	yes	1/4	1/16	5/16	.624
	5/8	6	1.5	0	yes	yes	3/8	5/16	11/16	1.09
	3/4	6	2	0	yes	yes	3/8	1/2	7/8	1.166
	7/8	6	2	0	yes	yes	3/4	9/16	1-5/16	1.50
	1	6	2.5	0	yes	yes	1	1/2	1-1/2	1.50
P-94	1/8 in.	6 in.	1 in.	1	yes	yes	3/16	1/8	5/16	2.50
	1/4	6	1	1	yes	yes	1/4	1/4	1/2	2.00
	3/8	6	1	1	yes	yes	1/4	1/2	3/4	2.00
	1/2	6	1.5	1	yes	yes	3/8	1/4	5/8	1.25
	5/8	6	1.5	1	yes	yes	1/2	5/16	13/16	1.30
	3/4	6	2	1	yes	yes	5/8	3/4	1-3/8	1.94
	7/8	6	2	1	yes	yes	7/8	3/4	1-5/8	1.86
	1	6	2.5	1	no	no			1-1/2	1.50
P-97	1/8 in.	6 in.	1 in.	1	yes	no	3/16	1/4	7/16	3.50
	1/4	6	1	1	yes	yes	1/4	1/4	1/2	2.00
	3/8	6	1	1	yes	yes	3/8	1/4	5/8	1.67
	1/2	6	1.5	1	yes	yes	1/2	3/4	1-1/4	2.50
	5/8	6	1.5	1	yes	yes	5/8	3/4	1-3/8	2.16
	3/4	6	2	1	no	yes	1/2	5/8	1-1/8	1.50
	7/8	6	2	1	yes	yes	3/4	3/4	1-1/2	1.50
	1	6	2.5	1	no	no				
P-103	1/8 in.	6 in.	1 in.	1	no	yes	3/16	1/8	5/16	2.50
	1/4	6	1	1	no	yes	1/4	1/4	1/2	2.00
	3/8	6	1	1	yes	yes	3/8	1/2	7/8	2.33
	1/2	6	1.5	1	no	yes	1/2	3/8	7/8	1.75
	5/8	6	1.5	1	yes	yes	1/2	1/2	1	1.60
	3/4	6	2	1	no	yes	1/2	1/2	1	1.33
	7/8	6	2	1	yes	yes	3/4	3/4	1-1/2	1.71
	1	6	2.5	1	no	no				
P-109	1/8 in.	6 in.	1 in.	2	yes	yes	3/16	1/8	5/16	2.50
	1/4	6	1	2	yes	yes	1/4	1/4	1/2	2.00
	3/8	6	1	2	yes	yes	1/4	1/4	1/2	1.33
	1/2	6	1.5	2	yes	yes	3/8	1/4	5/8	1.25
	5/8	6	1	2	yes	yes	1/2	1/2	1	1.60
	3/4	6	2	2	no	yes	3/4	3/4	1-1/2	2.00
	7/8	6	2	2	yes	yes	3/4	1	1-3/4	2.00
	1	6	2.5	2	no	no				
P-111	1/8 in.	6 in.	1 in.	2	yes	yes	1/4	1/4	1/2	2.00
	1/4	6	1	2	yes	yes	1/4	1/4	1/2	2.00
	3/8	6	1	2	yes	yes	1/4	1/4	1/2	1.33
	1/2	6	1.5	2	yes	yes	3/8	5/16	11/16	1.17
	5/8	6	1.5	2	yes	yes	1/2	1/4	3/4	1.20
	3/4	6	2	2	no	yes	1/2	1/2	1-3/8	2.00
	7/8	6	2	2	yes	yes	3/4	1/2	1-1/2	1.43
	1	6	2.5	2	no	no				

TABLE J12 (Continued)

HEAT NO.	PLATE			TAPER	UNDER RISER SHRINK	PLATE SHRINK	EDGE SOUND	SOUND BEYOND RISER	TOTAL SOUND	T =	
	THICK-NESS	DIA-METER	RISER DIA.							Total	Sound Thickness
P-113	1/8 in.	6 in.	1 in.	2							
	1/4	6	1	2	yes	yes	3/16	1/4	7/16		3.50
	3/8	6	1	2	yes	yes	1/4	1/4	1/2		2.00
	1/2	6	1.5	2	yes	yes	1/4	1/4	1/2		1.50
	5/8	6	1.5	2	no	yes	1/2	3/8	7/8		1.75
	3/4	6	2	2	yes	yes	5/8	1/4	7/8		1.40
	7/8	6	2	2	no	yes	7/8	3/4	1-3/8		1.83
	1	6	2.5	no	gas	yes	7/8	3/4	1-3/8		1.57
P-114	1/8 in.	6 in.	1.5 in.	3							
	1/4	6	1.5	3	no	yes	1/8	1/8	1/4		.20
	3/8	6	1.5	3	no	yes	1/4	1/4	1/2		2.00
	1/2	6	2	3	no	yes	3/8	1/4	5/8		1.67
	5/8	6	2.5	3	no	yes	1/2	1/4	3/4		1.50
	3/4	6	2.5	3	no	yes	1/2	1/2	1		1.804
	7/8	6	2.5	3	no	yes	1/2	3/8	7/8		1.167
	1	6	3	3	no	yes	3/4	3/4	1-1/2		1.71
P-118	1/8 in.	6 in.	1.5 in.	3							
	1/4	6	1.5	3	no	yes	1/8	1/8	1/4		2.00
	3/8	6	1.5	3	no	yes	3/8	1/2	7/8		3.50
	1/2	6	2	3	no	yes	1/2	1/2	1		2.67
	5/8	6	2	3	no	yes	3/8	1/2	7/8		1.75
	3/4	6	2	3	no	yes	5/8	1/2	1-1/8		1.85
	7/8	6	2.5	3	no	yes	3/4	3/4	1-1/2		2.00
	1	6	3	3	no	yes	3/4	3/4	1-1/2		1.71
P-121	1/8 in.	6 in.	1.5 in.	3							
	1/4	6	1.5	3	no	yes	1/8	1/4	3/8		3.00
	3/8	6	1.5	3	no	yes	3/8	1/2	7/8		2.86
	1/2	6	2	3	no	yes	1/2	1/2	1		2.66
	5/8	6	2	3	no	yes	1/2	1/2	1		2.00
	3/4	6	2.5	3	no	yes	1/2	1/2	1		1.60
	7/8	6	2.5	3	no	yes	5/8	3/4	1-5/8		2.17
	1	6	3.5	3	no	yes	3/4	1/2	1-1/4		1.43
P-123	1/8 in.	6 in.	1.5 in.	4							
	1/4	6	1.5	4	no	yes	1/4	1/4	1/2		4.00
	3/8	6	1.5	4	no	yes	3/8	1/4	7/8		2.86
	1/2	6	2	4	no	yes	1/2	1/2	1		2.66
	5/8	6	2	4	no	yes	5/8	1/2	1-1/8		2.50
	3/4	6	2.5	4	no	yes	1/2	1/4	3/4		1.20
	7/8	6	2.5	4	no	yes	5/8	1/2	1-1/8		1.50
	1	6	3.5	4	no	yes	1/2	7/8	1-3/8		1.57
P-124	1/8 in.	6 in.	1.5 in.	4							
	1/4	6	1.5	4	no	yes	1/4	3/8	5/8		5.00
	3/8	6	1.5	4	no	yes	3/8	1/2	7/8		2.86
	1/2	6	2	4	yes	yes	3/8	1/4	5/8		1.66
	5/8	6	2	4	no	yes	1/4	1/2	3/4		1.50
	3/4	6	2.5	4	no	yes	5/8	3/4	1-3/8		2.20
	7/8	6	2.5	4	no	yes	1/2	1/4	3/4		1.00
	1	6	3.5	4	no	no	1-3/4	1-3/4	1-3/4		2.00
P-125	1/8 in.	6 in.	1.5 in.	4							
	1/4	6	1.5	4	no	yes	1/8	3/8	1/2		4.00
	3/8	6	1.5	4	no	yes	1/4	3/8	5/8		2.50
	1/2	6	2	4	no	yes	1/4	1/2	3/4		2.00
	5/8	6	2	4	no	yes	1/2	1/2	1		2.00
	3/4	6	2.5	4	no	yes	1/2	3/4	1-1/4		2.00
	7/8	6	2.5	4	no	yes	5/8	1/2	1-1/8		1.50
	1	6	3	4	no	no	3/4	1/2	1-1/4		1.43

TABLE J12 (Continued)

HEAT NO.	PLATE			TAPER	UNDER RISER SHRINK	PLATE SHRINK	EDGE SOUND	SOUND BEYOND RISER	TOTAL SOUND	T =	
	THICK- NESS	DIA- METER	RISER DIA.							Total	Sound Thickness
P-126	1/8 in.	6 in.	1.5 in.	5	no	yes	1/4	1/4	1/2	4.00	
	1/4	6	1.5	5	no	yes	3/8	1/4	7/8	3.50	
	3/8	6	1.5	5	yes	yes	1/2	1/2	1	2.57	
	1/2	6	2	5	no	yes	3/4	5/8	1-3/8	2.75	
	5/8	6	2	5	no	yes	3/4	1	1-3/4	2.80	
	3/4	6	2.5	5	no	yes	3/4	3/4	1-1/2	1.71	
	7/8	6	2.5	5	no	yes	3/4	3/4	1-1/2	1.71	
P-127	1/8 in.	6 in.	1.5 in.	5	no	yes	1/4	1/4	1/2	4.00	
	1/4	6	1.5	5	no	yes	3/8	1/2	7/8	3.50	
	3/8	6	1.5	5	no	yes	1/2	7/8	1-3/8	3.67	
	1/2	6	2	5	no	yes	5/8	3/4	1-3/8	2.75	
	5/8	6	2	5	no	yes	3/4	3/4	1-1/2	2.40	
	3/4	6	2.5	5	no	yes	3/4	3/4	1-1/2	2.00	
	7/8	6	2.5	5	no	yes	3/4	3/4	1-1/2	2.00	
P-133	1/8 in.	6 in.	1.5 in.	5	no	yes	1/8	1/4	3/8	3.00	
	1/4	6	1.5	5	no	yes	3/8	1/2	5/8	2.50	
	3/8	6	1.5	5	no	yes	1/2	1/2	1	2.57	
	1/2	6	2	5	no	yes	5/8	1/2	1-1/8	2.50	
	5/8	6	2	5	no	yes	3/4	1/2	1-1/4	2.00	
	3/4	6	2.5	5	no	yes	3/4	3/4	1-1/2	2.00	
	7/8	6	2.5	5	no	yes	3/4	3/4	1-1/2	2.00	
P-134	1/8 in.	6 in.	1.5 in.	6	no	yes	1/8	1/2	5/8	5.00	
	1/4	6	1.5	6	no	yes	3/8	3/8	3/4	3.00	
	3/8	6	1.5	6	no	yes	1/2	1/2	1	2.67	
	1/2	6	2	6	no	yes	1/2	3/4	1-1/4	2.50	
	5/8	6	2	6	no	yes	5/8	1/2	1-1/8	1.80	
	3/4	6	2	6	no	yes	5/8	1/2	1-1/8	1.80	
	7/8	6	2.5	6	no	yes	5/8	1/2	1-1/8	1.80	
P-135	1/8 in.	6 in.	1.5 in.	6	no	yes	1/8	1/2	5/8	5.00	
	1/4	6	1.5	6	no	yes	3/8	3/8	3/4	3.00	
	3/8	6	1.5	6	yes	yes	1/2	5/8	1 1/8	3.31	
	1/2	6	2.0	6	no	yes	1/2	5/8	1 1/8	2.25	
	5/8	6	2	6	no	yes	3/4	3/4	1 1/2	2.41	
	3/4	6	2.5	6	no	yes	3/4	3/4	1 1/2	2.00	
	7/8	6	2.5	6	no	yes	1/2	3/4	1 1/4	1.71	
P-167	1/8 in.	6 in.	1.5 in.	6	no	yes	1/8	1/2	5/8	5.00	
	1/4	6	1.5	6	no	yes	3/8	3/8	3/4	3.00	
	3/8	6	1.5	6	yes	yes	1/2	5/8	1 1/8	3.31	
	1/2	6	2	6	no	yes	1/4	5/8	1 1/8	1.75	
	5/8	6	2	6	no	yes	1 1/2	1 1/4	2 3/4	4.40	
	3/4	6	2.5	6	no	yes	5/8	5/8	1 1/4	1.71	
	7/8	6	2.5	6	no	yes	5/8	5/8	1 1/4	1.71	
P-169	1/8 in.	6 in.	1.5 in.	7	no	yes	1/8	1/2	5/8	5.00	
	1/4	6	1.5	7	no	yes	1/2	1	1 1/2	6.00	
	3/8	6	1.5	7	yes	yes	3/4	3/4	1 1/2	4.60	
	1/2	6	2	7	no	yes	5/8	1	1 5/8	3.35	
	5/8	6	2	7	no	yes	5/8	1	1 5/8	3.35	
	3/4	6	2.5	7	no	yes	5/8	1	1 5/8	3.35	
	7/8	6	2.5	7	no	yes	5/8	1	1 5/8	3.35	
P-169	1/8 in.	6 in.	1.5 in.	7	no	yes	1/8	1/2	5/8	5.00	
	1/4	6	1.5	7	no	yes	1/2	1	1 1/2	6.00	
	3/8	6	1.5	7	yes	yes	1/2	1 1/2	1	2.67	
	1/2	6	2	7	no	yes	1/2	1 1/2	1	2.67	
	5/8	6	2	7	yes	yes	1/2	1 1/2	1	2.67	
	3/4	6	2.5	7	no	yes	1/2	1 1/2	1	2.67	
	7/8	6	2.5	7	yes	yes	1/2	1 1/2	1	2.67	

TABLE J12 (Continued)

Heat No.	PLATE				Under		Sound		Total Sound	To Total Sound Thickness
	Thick-ness	Dia-meter	Riser Dia.	Riser Taper	Riser Shrink	Plate Shrink	Edge Sound	Beyond Riser		
P171	1/8 in.	6 in.	1.5 in.	7	no	yes	3/8	3/4	1 1/8	9.00
	1/4	6	1.5	7	no	yes	5/8	1/2	1 1/8	4.50
	3/8	6	1.5	7	yes	yes	1/2	7/8	1 3/8	2.75
	1/2	6	2	7	no	yes	1/2	3/8	7/8	1.75
	5/8	6	2	7	yes	gas	-	-	-	-
	3/4	6	2.5	7	no	gas	-	-	-	-
	7/8	6	2.5	7	no	gas	-	-	-	-
	1	6	3	7	no	no	-	-	1 1/2	1.50
P172	1/8 in.	6 in.	1.5 in.	8	no	yes	3/8	3/4	1 1/8	9.00
	1/4	6	1.5	8	no	yes	3/8	1	1 3/8	5.50
	3/8	6	1.5	8	yes	yes	gas	1	1	2.67
	1/2	6	2	8	no	gas	-	-	-	-
	5/8	6	2	8	yes	gas	-	-	-	-
	3/4	6	2.5	8	no	gas	-	-	-	-
	7/8	6	2.5	8	no	gas	-	-	-	-
	1	6	3	8	no	gas	-	-	-	-
P173	1/8 in.	6 in.	1.5 in.	8	yes	yes	3/8	5/8	1	8.00
	1/4	6	1.5	8	no	yes	1/2	3/4	1 1/4	5.00
	3/8	6	1.5	8	yes	yes	3/8	5/8	1	2.67
	1/2	6	2	8	no	gas	-	-	-	-
	5/8	6	2	8	yes	yes	1/2	5/8	1 1/8	1.80
	3/4	6	2.5	8	no	gas	-	-	-	-
	7/8	6	2	8	no	gas	-	-	-	-
	1	6	3	8	no	gas	-	-	-	-
P174	1/8 in.	6 in.	1.5 in.	8	no	yes	1/2	1/2	1	8.00
	1/4	6	1.5	8	no	yes	1/2	1/2	1	4.00
	3/8	6	1.5	8	yes	yes	5/8	7/8	1 1/4	3.30
	1/2	6	2	8	no	gas	-	-	-	-
	5/8	6	2	8	yes	gas	-	-	-	-
	3/4	6	2.5	8	no	gas	-	-	-	-
	7/8	6	2.5	8	no	gas	-	-	-	-
	1	6	3	8	no	gas	-	-	-	-
P175	1/8	6	1.5	9	no	yes	gas	1	1	8.00
	1/4	6	1.5	9	no	yes	3/4	1 1/4	2	8.00
	3/8	6	1.5	9	yes	yes	1/2	1	1 1/2	4.00
	1/2	6	2	9	no	gas	-	-	-	-
	5/8	6	2	9	yes	no	-	-	2	3.20
	3/4	6	2.5	9	no	gas	-	-	-	-
	7/8	6	2.5	9	yes	gas	-	-	-	-
	1	6	3	9	no	gas	-	-	-	-
P184	1/8 in.	6 in.	1.5 in.	9	no	yes	gas	5/8	5/8	5.00
	1/4	6	1.5	9	no	yes	1/8	7/8	1	4.00
	3/8	6	1.5	9	yes	yes	1/8	3/4	7/8	2.00
	1/2	6	2	9	no	yes	1/2	7/8	1 3/8	2.75
	5/8	6	2	9	yes	yes	1/2	7/8	1 3/8	2.20
	3/4	6	2.5	9	no	no	-	-	1 3/4	2.34
	7/8	6	2.5	9	no	no	-	-	1 3/4	2.00
	1	6	3	9	no	gas	-	-	-	-
P185	1/8 in.	6 in.	1.5 in.	9	no	yes	5/8	7/8	1 1/2	10.00
	1/4	6	1.5	9	yes	yes	1/4	7/8	1 1/8	4.50
	3/8	6	1.5	9	yes	yes	1/2	1 1/8	1 5/8	4.35
	1/2	6	2	9	no	yes	1/4	5/8	7/8	1.75
	5/8	6	2	9	yes	no	-	-	2	3.20
	3/4	6	2.5	9	no	no	-	-	1 3/4	2.34
	7/8	6	2.5	9	no	gas	-	-	-	-
	1	6	3	9	no	gas	-	-	-	-

TABLE J12 (Continued)

Heat No.	Thick- ness	PLATE			Under		Sound		Total Sound	Total Sound	T ₀ Thickness
		Thick- ness	dia- meter	Riser Dia., Top	Riser Shrink	Plate Shrink	Edge Sound	Beyond Riser			
P195	1/8 in.	6	1.5 in.	10	no	gas	-	-	-	-	-
	1/4	6	1.5	10	yes	gas	-	-	-	-	-
	3/8	6	1.5	10	yes	gas	-	-	-	-	-
	1/2	6	2	10	no	gas	-	-	-	-	-
	5/8	6	2	10	yes	gas	-	-	-	-	-
	3/4	6	2.5	10	no	gas	-	-	-	-	-
	7/8	6	2.5	10	yes	gas	-	-	-	-	-
	1	6	3	10	no	gas	-	-	-	-	-
P196	1/8 in.	6	1.5 in.	10	no	yes	1/4	1 1/4	1 1/2	10.00	-
	1/4	6	1.5	10	yes	yes	2/8	1	1 3/8	5.50	-
	3/8	6	1.5	10	yes	yes	1/2	3/4	1 1/4	3.33	-
	1/2	6	2	10	no	yes	3/8	1	1 3/6	2.75	-
	5/8	6	2	10	no	gas	-	-	-	-	-
	3/4	6	2.5	10	no	no	-	-	1 3/4	2.33	-
	7/8	6	2.5	10	no	gas	-	-	-	-	-
	1	6	3	10	no	no	-	-	1 1/2	1.50	-
P200	1/8 in.	6	1.5 in.	10	no	yes	1/2	3/4	1 1/4	10.00	-
	1/4	6	1.5	10	yes	yes	1/4	1	1 1/4	6.67	-
	3/8	6	1.5	10	yes	yes	1/2	1 1/2	2	5.34	-
	1/2	6	2	10	no	-	gas	1 5/8	1 5/8	3.25	-
	5/8	6	2	10	yes	no	-	-	2	3.20	-
	3/4	6	2.5	10	no	-	gas	1 1/4	1 1/4	1.67	-
	7/8	6	2.5	10	no	no	-	-	1 1/2	1.72	-
	1	6	3	10	no	no	-	-	1 1/2	1.50	-
P275-1	1/8 in.	6	1.5 in.	11	no	yes	1/16	3/4	13/16	6.50	-
	1/4	6	2	11	no	yes	1/8	1/2	5/8	2.50	-
	3/8	6	2	11	no	no	-	-	2	5.34	-
	1/2	6	2	11	no	gas	-	-	-	-	-
P275-2	1/8 in.	6	1.5 in.	11	no	yes	1/4	1/2	3/4	6.00	-
	1/4	6	2	11	no	yes	3/8	1 1/4	1 5/8	6.50	-
	3/8	6	2	11	no	no	-	-	2	5.34	-
	1/2	6	2	11	yes	no	-	-	2	4.00	-
P278	1/8 in.	6	1.5 in.	11	yes	yes	1/8	7/8	1	8.00	-
	1/4	6	2	11	no	yes	1/4	3/4	1	4.00	-
	3/8	6	2	11	no	yes	3/8	1/4	1	2.66	-
	1/2	6	2	11	yes	gas	-	-	-	-	-
P279	1/8 in.	6	1.5 in.	12	no	yes	1/2	1/4	3/4	6.00	-
	1/4	6	2	12	no	yes	3/8	1 1/4	1 5/8	6.50	-
	3/8	6	2	12	no	gas	-	-	-	-	-
	1/2	6	2.5	12	no	no	-	-	-	3.50	-
P279	1/8 in.	6	1.5 in.	12	no	yes	1/8	1 1/4	1 3/8	11.00	-
	1/4	6	2	12	no	yes	1/8	1 1/4	1 3/8	5.50	-
	3/8	6	2	12	no	no	-	-	-	5.34	-
	1/2	6	2.5	12	no	no	-	-	-	3.50	-
P281	1/8 in.	6	1.5 in.	12	yes	yes	1/4	1	1 1/4	10.00	-
	1/4	6	2	12	no	gas	-	-	-	-	-
	3/8	6	2	12	no	gas	-	-	2	5.34	-
	1/2	6	2.5	12	no	no	3/4	1 1/4	1 1/2	3.75	-

TABLE J13

FEEDING DISTANCE IN HEATED MACHINED GRAPHITE MOLDS

Heat No.	Thick- ness	PLATE			Under Riser Shrink	Plate Shrink	Edge Sound	Sound Beyond Riser	Total Sound	Y ^a Total Sound Thickness
		Dia- meter	Riser Dia.	Tapar						
Ph00	1/8	6	1 1/2	0	no	yes	1/16	1/16	1/8	1.00
	1/4	6	1 1/2	0	no	yes	1/8	0	1/8	0.50
	3/8	6	1 1/2	0	no	yes	1/4	0	1/4	0.66
	1/2	6	2	0	no	yes	1/4	0	1/4	0.50
	5/8	6	2	0	no	yes	1/2	1/8	5/8	1.00
	3/4	6	2 1/2	0	no	yes	1/2	1/4	3/4	1.00
	7/8	6	2 1/2	0	no	yes	1/2	1/4	3/4	0.86
	1	6	3	0	no	yes	7/8	1/2	1 3/8	1.175
Ph02	1/8	6	1 1/2	0	no	yes	1/16	0	1/16	0.50
	1/4	6	1 1/2	0	no	yes	1/4	0	1/4	1.00
	3/8	6	1 1/2	0	no	yes	3/8	0	3/8	1.00
	1/2	6	2	0	no	yes	1/2	1/8	5/8	1.25
	5/8	6	2	0	no	yes	1/2	0	1/2	0.80
	3/4	6	2	0	no	yes	1/2	1/4	3/4	1.00
	7/8	6	2 1/2	0	no	yes	1/2	1/8	5/8	0.71
	1	6	3	0	no	yes	3/4	1/4	1	1.00
Ph10	1/8	6	1 1/2	0	no	yes	1/8	0	1/8	1.00
	1/4	6	1 1/2	0	no	yes	1/4	0	1/4	1.00
	3/8	6	1 1/2	0	no	yes	3/8	1/4	5/8	1.66
	1/2	6	2	0	no	yes	1/2	0	1/2	1.00
	5/8	6	2	0	no	yes	1/2	0	1/2	0.80
	3/4	6	2	0	no	yes	5/8	1/4	7/8	1.16
	7/8	6	2 1/2	0	no	yes	1/2	1/8	5/8	0.71
	1	6	3	0	no	yes	3/4	1/4	1	1.00
Ph14	1/8	6	1 1/2	1	no	yes	1/8	1/16	3/16	1.50
	1/4	6	1 1/2	1	no	yes	1/4	0	1/4	1.00
	3/8	6	1 1/2	1	no	yes	3/8	0	3/8	1.00
	1/2	6	2	1	no	yes	1/2	1/2	1	2.00
	5/8	6	2	1	no	yes	1/2	1/8	5/8	1.00
	3/4	6	2 1/2	1	no	yes	1/2	0	1/2	0.64
	7/8	6	2 1/2	1	no	yes	3/4	0	3/4	0.86
	1	6	3	1	no	yes	5/8	1/8	3/4	0.75
Ph36	1/8	6	1 1/2	1	no	yes	0	1/8	1/8	1.00
	1/4	6	1 1/2	1	no	yes	1/8	1/8	1/4	1.00
	3/8	6	1 1/2	1	yes	yes	3/8	1/8	1/2	1.33
	1/2	6	2	1	no	yes	3/8	1/4	5/8	1.25
	5/8	6	2	1	no	yes	1/2	0	1/2	0.80
	3/4	6	2 1/2	1	no	yes	1/2	1/4	3/4	1.00
	7/8	6	2 1/2	1	no	yes	3/4	0	3/4	0.86
	1	6	3	1	no	yes	7/8	1/2	1 1/2	1.375
Ph47	1/8	6	1 1/2	1	no	yes	1/8	0	1/8	1.00
	1/4	6	1 1/2	1	no	yes	1/8	0	1/8	0.50
	3/8	6	1 1/2	1	no	yes	1/4	1/16	5/16	0.83
	1/2	6	2	1	no	yes	3/8	0	3/8	0.75
	5/8	6	2	1	no	yes	1/2	0	1/2	0.80
	3/4	6	2 1/2	1	no	yes	5/8	1/4	7/8	1.17
	7/8	6	2 1/2	1	no	yes	1	1/4	1 1/4	1.63
	1	6	3	1	no	no	1 1/2	1/2	1 1/2	1.50

TABLE J13 (Continued)

Heat No.	PLATE				Under Riser Shrink	Plate Shrink	Edge Sound	Sound Beyond Riser	Total Sound	Total Sound Thickness
	Thick-ness	Dia-meter	Riser Dia.	Taper						
P477	1/8	6	1 1/2	3	no	yes	1/4	1/8	3/8	3.00
	1/4	6	1 1/2	3	no	yes	1/4	0	1/4	1.00
	3/8	6	1 1/2	3	no	yes	1/4	0	1/4	0.66
	1/2	6	2	3	no	yes	1/2	1/4	3/4	1.25
	5/8	6	2	3	yes	yes	1/2	0	1/2	0.80
	3/4	6	2 1/2	3	no	yes	1/2	1/2	1	1.33
	7/8	6	2 1/2	3	no	yes	1/2	1/2	1	1.14
	1	6	3	3	no	yes	1/2	1/2	1	1.00
P487	1/8	6	1 1/2	3	no	yes	1/8	0	1/8	1.00
	1/4	6	1 1/2	3	yes	yes	1/4	1/8	3/8	1.50
	3/8	6	1 1/2	3	yes	yes	1/4	3/8	5/8	1.66
	1/2	6	2	3	no	yes	1/8	1/4	5/8	1.25
	5/8	6	2	3	no	yes	1/2	1/4	3/4	1.20
	3/4	6	2 1/2	3	no	yes	1/2	3/8	7/8	1.16
	7/8	6	2 1/2	3	no	yes	3/4	1/4	1	1.14
	1	6	3	3	no	yes	5/8	1/2	1 1/8	1.125
P490	1/8	6	1 1/2	3	no	yes	1/8	0	1/8	1.00
	1/4	6	1 1/2	3	yes	yes	1/4	1/4	1/2	2.00
	3/8	6	1 1/2	3	yes	yes	3/8	1/8	1/2	1.33
	1/2	6	2	3	no	yes	1/2	0	1/2	1.00
	5/8	6	2	3	yes	yes	1/2	0	1/2	0.80
	3/4	6	2 1/2	3	no	yes	1/2	1/2	1	1.33
	7/8	6	2 1/2	3	no	yes	1/2	1/4	5/8	0.71
	1	6	3	3	no	yes	1	1/2	1 1/2	1.50
P502	1/8	6	1 1/2	5	no	yes	1/4	1/8	3/8	3.00
	1/4	6	1 1/2	5	no	yes	1/4	1/8	3/8	1.50
	3/8	6	1 1/2	5	no	yes	1/4	5/8	4/8	1.66
	1/2	6	2	5	no	yes	1/2	1/4	3/4	1.50
	5/8	6	2	5	yes	yes	1/2	1/8	5/8	1.00
	3/4	6	2	5	no	yes	1/8	1/4	3/8	0.50
	7/8	6	2	5	yes	yes	3/4	1/4	1	1.14
	1	6	3	5	no	yes	1	3/8	1 5/8	1.375
P504	1/8	6	1 1/2	5	no	yes	1/16	1/8	3/16	1.50
	1/4	6	1 1/2	5	no	yes	1/4	1/8	3/8	1.50
	3/8	6	1 1/2	5	no	yes	1/2	1/2	1	2.67
	1/2	6	1 1/2	5	yes	yes	1/2	1/2	1	2.00
	5/8	6	2	5	yes	yes	3/4	1/8	7/8	1.50
	3/4	6	2	5	no	yes	3/4	1/2	1 1/4	1.67
	7/8	6	2 1/2	5	yes	yes	3/8	1/4	1	1.20
	1	6	3	5	no	yes	1/4	1/2	1 1/4	1.25
P505	1/8	6	1 1/2	5	no	yes	1/4	1/16	5/16	2.50
	1/4	6	1 1/2	5	no	yes	1/4	0	1/4	0.50
	3/8	6	1 1/2	5	no	yes	1/4	0	1/4	0.30
	1/2	6	2	5	no	yes	3/4	0	3/4	0.75
	5/8	6	2	5	yes	yes	1/8	1/4	3/8	1.00
	3/4	6	2 1/2	5	no	yes	1	1/4	1 1/4	1.75
	7/8	6	2 1/2	5	yes	yes	1/4	3/4	1	1.04
	1	6	3	5	no	yes	1	1/2	1 1/2	1.50

TABLE J13 (Continued)

Heat No.	Thick-ness	PLATE			Under Riser Shrink	Plate Shrink	Edge Sound	Sound Beyond Riser	Total Sound	T = Total Sound Thickness
		Dia-meter	Riser Dia.	Taper						
P506	1/8	6	1 1/2	7	yes	yes	1/4	0	1/4	2.00
	1/4	6	1 1/2	7	no	yes	1/4	1/2	3/4	3.00
	3/8	6	1 1/2	7	no	yes	1/2	3/4	1 1/4	3.00
	1/2	6	2 1/2	7	no	yes	1/2	1/4	3/4	1.50
	5/8	6	2 1/2	7	no	yes	1	1/2	1 1/2	2.40
	3/4	6	2 1/2	7	no	yes	3/4	1/4	1	1.33
	7/8	6	3	7	no	yes	5/8	1/2	1 1/8	1.28
	1	6	3	7	no	no	--	--	1 1/2	1.50
P507	1/8	6	1	7	no	yes	1/4	1/4	1/2	4.00
	1/4	6	1 1/2	7	no	yes	1/4	1/4	1/2	2.00
	3/8	6	1 1/2	7	no	yes	1/2	3/4	1 1/4	3.00
	1/2	6	2	7	no	yes	1/2	1/4	3/4	1.50
	5/8	6	2	7	no	yes	3/4	1/2	1 1/4	2.00
	3/4	6	2 1/2	7	no	yes	1	1/2	1 1/2	2.00
	7/8	6	2 1/2	7	no	yes	1	1/2	1 1/2	1.73
	1	6	3	7	no	no	--	--	1 1/2	1.50
P508	1/8	6	1 1/2	7	no	yes	1/8	0	1/8	1.00
	1/4	6	1 1/2	7	no	yes	1/4	1/8	3/8	1.50
	3/8	6	1 1/2	7	yes	yes	3/8	1/4	5/8	1.67
	1/2	6	2	7	no	yes	1/2	3/4	1 1/4	2.50
	5/8	6	2	7	no	yes	1/2	1	1 1/2	2.40
	3/4	6	2 1/2	7	no	yes	3/4	3/8	1 1/8	1.50
	7/8	6	2 1/2	7	no	yes	7/8	1/2	1 3/8	1.64
	1	6	3	7	no	no	--	--	1 1/2	1.50

TABLE J14

FEEDING DISTANCE IN HEATED RAMMED GRAPHITE MOLDS

Heat No.	Thick-ness	PLATE			Under		Edge Sound	Sound		Total Sound	T* Total Sound Thickness
		Dia-Meter	Riser Dia.	Taper	Riser Shrink	Plate Shrink		Beyond Riser			
P383	1/8	6	1	0	no	yes	(didn't fill)	0	0	0	0
	1/4	6	1	0	yes	yes	1/8	3/16	5/16	1.250	
	3/8	6	1	0	yes	yes	1/4	3/16	7/16	1.165	
	1/2	6	1 1/2	0	no	yes	1/8	1/4	3/8	.750	
	5/8	6	1 1/2	0	yes	yes	3/8	3/4	1 1/8	1.80	
	3/4	6	2	0	no	yes	1/2	1/2	1	1.33	
	7/8	6	2	0	no	yes	5/8	1 1/8	1 3/4	2.00	
	1	6	2 1/2	0	no	yes	5/8	3/4	1 3/8	1.375	
P384	1/8	6	1	0	yes	yes	1/16	1/8	3/16	1.50	
	1/4	6	1	0	yes	yes	1/8	3/8	1/2	2.00	
	3/8	6	1	0	yes	yes	1/4	1/4	1/2	1.33	
	1/2	6	1 1/2	0	yes	yes	1/8 gas	5/8	3/4	1.50	
	5/8	6	1 1/2	0	yes	yes	3/8	5/8	1	1.60	
	3/4	6	2	0	no	yes	1/2	1/2	1	1.33	
	7/8	6	2	0	yes	yes	1/2	7/8	1 3/8	1.57	
	1	6	2 1/2	0	no	no			1 3/4	1.750	
P385	1/8	6	1	0	yes	yes	1/32	0	1/32	.276	
	1/4	6	1	0	yes	yes	1/8	1/4	3/8	1.50	
	3/8	6	1	0	yes	yes	1/4	1/4	1	2.67	
	1/2	6	1 1/2	0	no	yes	1/4	1/4	1	2.00	
	5/8	6	1 1/2	0	yes	yes	3/8	1/2	7/8	1.40	
	3/4	6	1 1/2	0	no	yes	1/2	3/4	1 1/4	1.67	
	7/8	6	2	0	yes	yes	1/2	1 1/8	1 5/8	1.86	
	1	6	2 1/2	0	no	no			1 3/4	1.75	
P386	1/8	6	1	1	yes	yes	0 g	0	0	0	
	1/4	6	1	1	yes	yes	1/16 g	1/4	5/16	1.25	
	3/8	6	1	1	yes	yes	1/4 g	3/8	5/8	1.66	
	1/2	6	1 1/2	1	yes	yes	3/8 g	1/2	7/8	1.75	
	5/8	6	1 1/2	1	yes	yes	3/8	1/2	7/8	1.40	
	3/4	6	2	1	no	yes	1/2	5/8	1 1/8	1.50	
	7/8	6	2	1	no	yes	1/2	1	1 1/2	1.71	
	1	6	3	1	no	no			1 1/2	1.50	
P387	1/8	6	1	1	yes	yes	0 gas	1/8	1/8	1.00	
	1/4	6	1	1	yes	yes	1/8 g	1/8	1/4	1.00	
	3/8	6	1	1	yes	yes	1/4	1/4	1	2.67	
	1/2	6	1 1/2	1	yes	yes	1/4	1/2	3/4	1.50	
	5/8	6	1 1/2	1	yes	yes	1/4	1/4	1	1.60	
	3/4	6	2	1	no	yes	5/8	1/2	1 1/8	1.50	
	7/8	6	2	1	yes	no		3/4	1 3/4	2.00	
	1	6	2 1/2	1	no	no			1 3/4	1.75	

TABLE J14 (Continued)

FEEDING DISTANCE IN HEATED RAMMED GRAPHITE MOLDS

Heat No.	Thick-ness	PLATE			Under Riser Shrink	Plate Shrink	Edge Sound	Sound Beyond Riser	Total Sound	T = Total Sound Thickness
		Dia- Meter	Riser Dia.	Taper						
P388	1/8	6	1	1	yes	yes	0	0	0	0
	1/4	6	1	1	yes	yes	1/8 g	1/2	5/8	1.50
	3/8	6	1	1	yes	yes	1/4	1/4	1	2.67
	1/2	6	1 1/2	1	yes	yes	1/4	1/2	3/4	1.50
	5/8	6	1 1/2	1	yes	yes	1/2	1/2	1	1.60
	3/4	6	2	1	no	yes	1/2	1/2	1	1.33
	7/8	6	2	1	yes	yes	1/2	3/4	1 1/4	1.43
	1	6	2 1/2	1	no	no	gas	1	3/4	1.750
P389	1/8	6	1 1/2	3	no	yes	gas	1/4	1/4	2.00
	1/4	6	1 1/2	3	no	yes	gas	1/8	1/8	.50
	3/8	6	1 1/2	3	no	yes	1/4 g	1/4	1	2.67
	1/2	6	2	3	no	yes	3/8 g	3/4	1 1/8	2.25
	5/8	6	2	3	no	yes	3/8 g	1/2	7/8	1.40
	3/4	6	2 1/2	3	no	no	gas	1	3/4	2.34
	7/8	6	2 1/2	3	no	no	gas	1	3/4	2.00
	1	6	3	3	no	no		1	1/2	1.50
P390	1/8	6	1 1/2	3	no	yes	gas	1/4	1/4	2.00
	1/4	6	1 1/2	3	no	yes	1/4 g	1/4	1/4	1.00
	3/8	6	1 1/2	3	no	yes	1/8 g	1/8	1/4	1.667
	1/2	6	2	3	no	yes	3/8	1/2	7/8	1.750
	5/8	6	2	3	no	yes	1/2	1/2	1	1.60
	3/4	6	2 1/2	3	no	yes	5/8	1	5/8	2.17
	7/8	6	2 1/2	3	no	no	gas	1	3/4	2.00
	1	6	3	3	no	no	gas	1	1/2	1.50
P391	1/8	6	1 1/2	3	no	yes	gas	gas		0
	1/4	6	1 1/2	3	no	yes	1/4 g	1/8	3/8	1.50
	3/8	9	1 1/2	3	no	yes	1/4	1/4	1/2	1.33
	1/2	6	2	3	no	yes	1/4	1/2	3/4	1.50
	5/8	6	2	3	no	yes	1/2	3/4	1 1/4	2.00
	3/4	6	2 1/2	3	no	yes	1/2	1	1 1/2	2.00
	7/8	6	2 1/2	3	no	no	gas	1	3/4	2.00
	1	6	3	3	no	no	gas	1	1/2	1.50
P392	1/8	6	1 1/2	5	no	yes	1/2 g	1/2	1	8.00
	1/4	6	1 1/2	5	no	yes	1/4 g	1/4	1/2	2.00
	3/8	6	1 1/2	5	no	yes	1/2	5/8	1 1/8	3.00
	1/2	6	2	5	no	yes	1/2	3/4	1 1/4	2.50
	5/8	6	2	5	no	yes	1/2	1/2	1	1.60
	3/4	6	2 1/2	5	no	no	gas	1	3/4	2.80
	7/8	6	2 1/2	5	no	no	gas	1	3/4	2.34
	1	6	3	5	no	no	gas	1	1/2	1.50

TABLE J14 (Continued)

FEEDING DISTANCE IN HEATED RAMMED GRAPHITE MOLDS

Heat No.	Thick-ness	PLATE			Under Riser Shrink	Plate Shrink	Edge Sound	Sound Beyond Riser	Total Sound	T=	
		Dia-Meter	Riser Dia.	Taper						Total Sound	Thickness
P393	1/8	6	1 1/2	5	no	yes	0 g	1/4	1/4	4.00	
	1/4	6	1 1/2	5	no	yes	1/4 g	3/8	5/8	2.50	
	3/8	6	1 1/2	5	yes	yes	1/4 g	1/4	1/2	1.33	
	1/2	6	2	5	no	yes	3/8	3/4	1 1/8	2.25	
	5/8	6	2	5	no	yes	1/2	3/4	1 1/4	2.00	
	3/4	6	2 1/2	5	no	yes	1/2	1	1 1/2	2.00	
	7/8	6	2 1/2	5	no	no			1 3/4	2.00	
	1	6	3	5	no	no			1 1/2	1.50	
P394	1/8	6	1 1/2	5	no	yes	gas	1/2	1/2	4.00	
	1/4	6	1 1/2	5	no	yes	1/4 g	1/2	3/4	3.00	
	3/8	6	1 1/2	5	yes	yes	1/4	1/4 g	1/2	1.37	
	1/2	6	2	5	no	yes	1/2 g	3/4	1 1/4	2.50	
	5/8	6	2	5	no	yes	7/8	3/4	1 5/8	2.60	
	3/4	6	2 1/2	5	no	no			1 3/4	2.34	
	7/8	6	2 1/2	5	no	no			1 3/4	2.00	
	1	6	3	5	no	no			1 1/2	1.50	
P395	1/8	6	1 1/2	7	no	yes	gas	0	0		
	1/4	6	1 1/2	7	no	yes	1/8	1 3/4	1 7/8	7.50	
	3/8	6	1 1/2	7	no	yes	1/4	3/4	1	2.67	
	1/2	6	2	7	no	gas	3/16	1/4	7/16	.875	
	5/8	6	2	7	yes	gas	1/8	3/4	7/8	1.40	
	3/4	6	2 1/2	7	no	gas	1/4		2	2.67	
	7/8	6	2 1/2	7	no	no			1 3/4	2.00	
	1	6	3	7	no	no			1 1/2	1.50	
P396	1/8	6	1 1/2	7	no	yes	1/16	3/8	7/16	3.50	
	1/4	6	1 1/2	7	no	yes	0	1/2	1/2	2.00	
	3/8	6	1 1/2	7	yes	yes	1/8	1/2	5/8	1.67	
	1/2	6	2	7	no	yes	1/2	1/4	3/4	1.50	
	5/8	6	2	7	no	gas					
	3/4	6	2 1/2	7	no	gas					
	7/8	6	2 1/2	7	no	gas					
	1	6	3	7	no	gas					
P397	1/8	6	1 1/2	7	no	yes	0	1/4	1/4	2.00	
	1/4	6	1 1/2	7	no	yes	0	3/4	3/4	3.00	
	3/8	6	1 1/2	7	yes	yes	1/2	3/4	1 1/4	3.34	
	1/2	6	2	7	no	gas					
	5/8	6	2	7	no	no			2	3.20	
	3/4	6	2 1/2	7	no	gas					
	7/8	6	2 1/2	7	no	no			1 3/4	2.00	
	1	6	3	7	no	no			1 1/2	1.50	

TABLE J15

FEEDING DISTANCE IN CENTRIFUGED DISCS

HEAT NO.	PLATE		RISER SIZE	EDGE SOUND	SOUND BEYOND RISER	TOTAL SOUND	T = Total Sound Thickness	RPM/G'S
	THICKNESS	DIAMETER						
P-80	1/2	6	1	3/16	1/4	7/16	0.875	900/200
	1/2	6	1	1/4	1/4	1/2	1.000	
	1/2	6	1	3/16	1/4	7/16	0.875	
	1/2	6	1	3/16	1/8	5/16	0.630	
P-81	1/2	6	1	1/4	5/8	7/8	1.750	900/200
	1/2	6	1	3/16	1/4	7/16	0.875	
	1/2	6	1	3/16	5/16	1/2	1.000	
	1/2	6	1	3/16	1/4	7/16	0.875	
P-83	1/2	6	1	1/4	5/8	7/8	1.750	900/200
	1/2	6	1	3/16	1/4	7/16	0.875	
	1/2	6	1	3/16	1/4	7/16	0.875	
	1/2	6	1	3/16	5/8	13/16	1.625	
P-85	1/2	6	1 1/2	Gas	5/8	5/8	1.250	900/200
	1/2	6	1 1/2	0	1/4	1/4	0.500	
	1/2	6	1 1/2	Gas	5/16	5/16	0.630	
	1/2	6	1 1/2	Gas	1/2	1/2	1.000	
P-86	1/2	6	1 1/2	1/4	1/4	1/2	1.000	900/200
	1/2	6	1 1/2	3/16	1/4	7/16	0.875	
	1/2	6	1 1/2	3/16	1/4	7/16	0.875	
	1/2	6	1 1/2	1/4	1/4	1/2	1.000	
P-99	1/2	6	1 1/2	3/16	1/4	7/16	0.875	1300/325
	1/2	6	1 1/2	1/4	1/4	1/2	1.000	
	1/2	6	1 1/2	1/4	1/4	1/2	1.000	
	1/2	6	1 1/2	1/4	1/4	1/2	1.000	

TABLE J16

RESULTS OF TENSION FATIGUE TESTS TO ESTABLISH CHEMICAL REMOVAL REQUIREMENTS

SPECIMEN CONDITION	MAXIMUM STRESS, PSI	CYCLES TO FAILURE
As Cast and Abrasive Blast Cleaned	75,000	101,000
	75,000	82,000
	75,000	92,000
	75,000	151,000
	65,000	<u>1,165,000</u>
	60,000	<u>1,546,000</u>
	60,000	<u>1,024,000</u>
	60,000	<u>129,000</u>
.005" Per Surface Chemically Removed	75,000	61,000
	75,000	114,000
	75,000	33,000
	75,000	88,000
	65,000	66,000
	65,000	177,000
	65,000	205,000
	55,000	<u>1,026,000</u>
.010" Per Surface Chemically Removed	75,000	83,000
	75,000	74,000
	75,000	127,000
	75,000	213,000
	65,000	170,000
	65,000	182,000
	65,000	95,000
	65,000	<u>1,072,000</u>
.015" Per Surface Chemically Removed	85,000	724,000
	85,000	251,000
	75,000	297,000
	75,000	185,000
	75,000	744,000
	75,000	<u>1,013,000</u>
	65,000	<u>1,252,000</u>
	65,000	<u>1,561,000</u>

Underlined values were not failures.

$$R = \frac{\text{Minimum Stress}}{\text{Maximum Stress}} = .06$$

TABLE J17

MECHANICAL PROPERTIES OF AS-CAST Ti-6Al-4V

Heat Number	Ultimate Tensile Strength KSI	Yield Strength KSI (.2%)	Elongation in 4D Per Cent	Reduction in Area Per Cent	Notch Tensile Strength KSI	Brinell Hardness Number
P2-7	119.8	108*	4	9.2		
-7	152.3	127.1	12	22.6		
P4-8	156.1	not obtained	11	22.6		
-8	140.1	110*	10.5	15.5		
P6-5A					216.85	
-5B					210.4	
-5C					216.85	
-6	141.6	124.7	12	24		
P7-4	144.85	128.4	9	18		
	144.5	128.2	11	21		
	143.7	130.9	9.5	12		
	144.85	128.4	9	18		
	144.5	128.2	11	21		
	143.7	130.9	9.5	12		
P7-2G					213.55	
-2H					215.15	
-2I					215.2	
P9-2	145.9	131.35	9	12		
	145.55	131.4	9.5	19		
	144.9	132.5	9.5	18		
P10-2	150.6	131	10	20		
P11-2	144.5	120.5	8.5	17		
P12-1	148.65	136.15	6.5	10**		
P13-1	144.5	133.65	10	18		
P14-1	148.15	136.05	10	15		
P15-1	138.25	127.15	10	25		
P16-1	143.9	131.65	8	18		
P17-1	141.6	126.7	5	6		
P18-2	149	130.5	8	13		
P19-5	130.9	112.6	10	27 (off analysis)		
P20-1	143.0	120	12	21		
P23-1	144.8	132.3	9	18		
P27-2B	147	135.5	9	15		
P29-2	149	Poor Curve	9	10		
P30-2	150.5	137.25	9	15		
P31-6	147.2	130.1	7	20		
P32-2	149.9	133.2	5	7		
P34-2	151.45	138.25	9.5	18		
P35-2	144.5	127.35	13	23		
P36-2	150.2	133.3	5	13		
P37-2	151	133.3	6	15		
P40-3	153.1	142	8.5	16		
P41-2	152.9	145.7	8	17		
P42-2	146.1	132.8	13	20		

*Flaw at fracture

**Broke out of gage length

TABLE J17 (Continued)

MECHANICAL PROPERTIES OF AS-CAST Ti-6Al-4V

Heat Number	Ultimate Tensile Strength KSI	Yield Strength KSI (.2%)	Elongation in 4D Per Cent	Reduction In Area Per Cent	Notch Tensile Strength KSI	Brinell Hardness Number
P43-2	151.2	140.9	6.5	13		
P45-2	145.1	130.8	11	22		
P47-2	154.7	143.5	12	20		
P48-2	154.3	141.8	12.5	18		
P50-2	150	134.95	12	18		
P51-2	152.8	137.15	12.5	18.5		334
P53-2	152.6	139.1	8	15		324
P54-2	152.9	143.2	9.5	14		324
P56-2	145.5	129.4	12	25		321
P57-3	144	128.8	10	19		321
P63-2	149.2	135.8	10	16		321
P70-2	148.9	135.7	15	23		337
P71-2	151.9	138.3	8	12		
P73-3	156.4	143.2	12	18		
P76-2	138.4	124	8	20.4		317
P78-2	156.4	138	7	16		343
P100-2	154.8	137.6	7	17		348
P101-1	157.2	121	9	15.1		345
P102-2	142.8	125.2	9	22		345
P104-2	146.8	130	9	22		348
P108-3	144	127.6	7	19.4		337
P115-3	144	127.2	9	26.5		317
P119-2	146.4	128.4	8	19.4		344
P129-2	151.6	135	8	24.1		334
P131-2	142.8	124.4	9	18.9		317
P132-2	148	143	7	18.1		338
P137-2	144	125	6	15.1		314
P142-2	156.8	135	8	17.6		344
P155-2	144	126	3	10		308
P156-2	152.4	138	8	14.2		324
P157-2	149.2	132	7	13.4		317
P158-1	141.6	124	8	17		317
P159-2	145.6	128	9	15.1		311
P186-2	139.4	118	8	9.7		302
P194-1	142	124	7	11.5		305
P197-1	132	116	7	13.1		285
P198-1	142	124	6	10.5		321
P199-2	144	125	8	15.1		311
P201-2	140	120	7	12.5		305
P206-3	138.4	121	8	15.9		299
P210-1	148	131	9	14.5		311
P211-1	142	124	9	15.1		305
P217-1	145.2	127	8	10		296
P218-1	144.2	126	7	16.7		314.3
P220-1	148	129	8	14.5		305
P221-1	136	118	4	7.6		299
P224-1	146	126	6	10.5		308.3
P225-1	150	132	7	10		331
P232-1	148	121	6	11.2		308

TABLE J17 (Continued)

MECHANICAL PROPERTIES OF AS-CAST TI-6Al-4V

Heat Number	Ultimate Tensile Strength KSI	Yield Strength KSI (.2%)	Elongation in 4D Per Cent	Reduction in Area Per Cent	Notch Tensile Strength KSI	Brinell Hardness Number
P240-1	148	130	8	12.9		308
P241-1	142.2	125	8	13.5		299
P242-1	139.2	122	8	12.9		296
P243-1	139.2	112	5	14.5		311
P244-2	141.2	124	5	13.4		311.6
P246-1	139.6	121	8	15		311.3
P247-1	145.2	128	8	14.5		305
P248-1	144	126	8	14.5		308
P250-1	140	122	7	11.5		293.6
P260-1	146	127	6	8		296
P261-1	146	129	8	13.5		317.6
P264-1	146	130	6	4.9		321
P265-1	143.2	126	4	9.7		311
P281-1	144	126	10	15		316
P284-1	144	126	7	18.1		308
P286-1	138.4	121	7	12.7		308
P290-1	149.6	132	9	19.4		317.6
P291-1	138	120	9	17.5		311.3
P304-1	146.2	131	7	13.5		321
P305-1	148	131	7	10		324.3
P306-1	146	128	7	10		308
P307-1	146.8	130	8	15.9		305
P308-1	146.8	130	6	11.5		317.6
P309-1	152	134	7	13.5		317.6
P312-2	146.8	132	8	14.2		321
P314-1	146.4	130	8	12.9		336
P322-1	146	129	9	15.1		331
P323-1	144.8	128	9	13.4		321
P324-1	152	135	7	12.7		326
P328-1	142	124	9	12.7		311
P329-1	142	124	7	8.1		311
P330-2	148	132	9	14.5		321
P333-1	140.4	122	9	15.9		302
P336-1	144	137	8	15.1		326
P340-1	145.2	130	7	12.7		327.2
P343-1	150	135	7	16.7		324.3
P345-3	143.8	128	6	8.9		317.6
P346-2	140.2	122	7	15.1		311
P360-1	149.6	132	7	10.5		327.2
P364-1	146.8	131	7	13.5		305
P365-1	150	134	7	12.2		311
P366-1	149.6	128	8	15.1		311
P367-1	150	134	6	10.6		311
P368-1	153.6	136	6	10.6		327.6
P369-1	148.4	133	7	12.9		317.6
P370-1	150.8	134	7	12.7		314.3
P371	146.8	131	7	8.9		331
P373	151.2	132	7	12.2		324.3
P374	150.8	135	5	10.6		317.6
P379	156.0	139	7	9.2		334.3

TABLE J17 (Continued)

MECHANICAL PROPERTIES OF AS-CAST Ti-6Al-4V

Heat Number	Ultimate Tensile Strength KSI	Yield Strength KSI (.2%)	Elongation In 4D Per Cent	Reduction In Area Per Cent	Notch Tensile Strength KSI	Brinell Hardness Number
P380-1	154	137	7	8.6		331
P381-1	151.6	134	7	10		331
P382-1	152	135	7	14.5		331
P398-1	150.4	134	7	11.5		321
P399-1	140.4	123	7	12.9		302
P401-1	150	130	6	12.9		311
P403-1	150	133	5	10.5		321
P404-1	148	132	6	14.5		311.3
P405-1	154.8	137	8	12.2		327.6
P408-1	150	134	7	13.5		327.6
P411-1	147.2	129	4	10.5		324.4
P412-1	150	132	6	15.1		331
P413-1	139.6	120	8	16.7		314.3
P414-1	150	132	8	16.7		317.6
P418-1	146.8	128	8	16.7		321
P419-1	153.6	135	6	13.5		337.6
P420-1	142	122	8	21.7		321
P421-1	141.6	122	8	14.2		311
P422-1	152	133	9	18.1		324.3
P423-1	146	130	9	17.5		308
P511-1	146	131	8	12.9		311
P512-1	152	134	7	13.5		324.3
P524	153.2	137	7	13.5		331
P525	144.4	128	8	18.1		---
P526	153.6	135	7	13.5		331
P528	147.2	131	7	11.5		311.3
P529	149.2	132	8	15.1		324.3
P530	147.2	130	8	11.2		324
P531	148.2	130	7	12.2		317.6
P532	147.2	129	9	19.4		327.3
P534	150	132	7	18.9		331
P535	146.4	128	7	17.5		331
P536	152	132	8	10.6		
P538	152.8	133	9	15.1		---
P539	150	132	7	18.1		321
P540	147.2	130	6	11.5		314.4
P541-1	148.4	132	8	13.4		321
P542-1	146.8	130	7	18.1		305
P543-1	150	133	6	12.7		327.3
P546-1	143.6	126	5	12.2		306.5
P548-1	150	134	7	13.4		316
P549	165.2	145	8	13.5		311
P551	145.6	128	7	14.5		316
P552	146.2	130	8	15.9		331
P554	146.2	129	8	14.5		316
P555	148	129	10	23.7		302
P557	148.8	131	11	21.7		302
P558-1	148	129	8	17.5		292
P559	149.2	131	9	17.8		363
P562	155.2	136	9	17.5		311
P560	154	134	13	22.4		331
P561	154.8	125	11	16.7		265

TABLE J17 (Continued)

MECHANICAL PROPERTIES OF AS-CAST Ti-6Al-4V

Heat Number	Ultimate Tensile Strength KSI	Yield Strength KSI (.2%)	Elongation in 40 Per Cent	Reduction in Area Per Cent	Notch Tensile Strength KSI	Brinell Hardness Number
P564	144.8	130	9	16.4		
P566-1	152.4	136	6	10.5		
P568-1	143.6	128	8	17.0		
P575-1	150	134	7	15.9		
P570	143.2	126	7	16.4		
P571	146.8	129	8	15.9		
P572	143.6	126	8	14.2		
P573	146.4	130	6	16.4		
P574	145.2	130	5	12.2		
P576	147.2	131	7	11.2		
P577-1	142.8	122	13	27		302
P578-1	148.8	128	9	19.4		311
P580-1	146.4	128	11	21		311
P581-1	144	124	10	20.4		302
P582-1	135.6	118	10	18.1		
P584-1	146.8	124	10	19.7		302
P585-1	158	138	9	19.7		331
P586-1	146	128	8	15.8		302
P586-1A	148.8	130	9	10		302
P586-1A	146.4	128	9	18.1		302
P586-2A	146.8	128	9	20.5		293
P587-1	158.8	137	7	13.5		321
P588-1	152	132	9	16.7		311
P589-1	152.4	129	9	15.9		
P590-1	160	138	10	21		
P591-1	154.8	134	8	15.1		
P592-1	168.8	146	5	10.5		
P593-1	154.8	134	11	20.8		
P595-1	156.4	136	8	15.9		
P598-1	162.4	140	8	15.1		
P599-1	156	136	10	15.9		
P604-1	150	130	8	16.7		311
P608-3	151.2	132	9	13.5		331
P609-1	152.8	136	9	18.1		
P610-1	144.8	123	9	18.1		
P611-1	150.4	130	9	21		
P612-1	150.8	132	11	24.8		
P613-1	149.2	128	9	17.5		
P614-1	151.2	128	9	18.3		
P617-1	148.8	130	11	24.6		
P618-1	148	128	11	23.7		
P619-1	142.2	120	11	24.6		
P620-1	141.8	120	11	23.7		
P621-3	151.2	129	10	20.6		
P622-3	147.2	123	13	26.1		
P623-3	147.8	125	11	21.7		
P624-3	154.8	132	9	17.5		
P625-3	150	129	12	27.2		

TABLE J18

ELEVATED TEMPERATURE PROPERTIES OF AS-CAST Ti-6Al-4V

HEAT NO.	TEST TEMPERATURE, °F	ULTIMATE TENSILE STRENGTH, KSI	YIELD STRENGTH, KSI (.2%)	ELONGATION, PER CENT	REDUCTION, OF AREA, PER CENT	MODULUS OF ELASTICITY, PSI $\times 10^{-6}$
467	70	140.5	127.8	8	13	16.5
467	70	141.0	129.6	11	23	16.9
469	70	141.5	126.3	10	13	16.0
469	70	141.2	126.6	10	15	16.9
467	400	107.2	88.5	12	20	14.8
467	400	107.8	90.6	12	24	16.9
469	400	107.2	90.5	9	10	15.7
469	400	107.6	91.9	10	16	15.7
467	600	94.3	75.7	11	24	14.3
467	600	92.7	73.5	12	27	13.7
469	600	90.6	72.9	11	23	13.5
469	600	92.0	74.2	11	23	15.6
467	800	84.8	65.3	11	34	12.8
467	800	85.8	64.0	12	31	12.0
469	800	82.9	66.0	13	37	13.0
469	800	82.4	65.3	11	34	12.9

NOTE: The first two tests at each temperature were from heat 467.

TABLE J19

MECHANICAL PROPERTIES OF HEAT TREATED CAST Ti-6Al-4V

Heat No.	Heat Treatment	Yield Strength KSI (44)	Ultimate Tensile Strength KSI	Elongation In 40 Per Cent	Reduction In Area Per Cent	Notch Tensile KSI	Brinell Hardness Number
P7	As Cast	145.1	124.1	8	19.0		
"	1900F-1/2HR-WQ +1750-1/2HR-WQ+1000F-4HR-AC	168.1	156.7	3	5.0		
P10	1900F-1/2HR-WQ +1750F-1/2HR-WQ " " "	174.1	162.6	2	5.0		
"	1750F-1/2HR-WQ " " "	173.2	162.7	4	1.0		
"	As Cast	156.8	140.6	7	19.0		
P11	As Cast	146.8	127.2	6	14.0		
"	1900F-1/2HR-WQ+1750F-1/2HR-WQ +1750-1/2HR-WQ+1000F-4HR-AC	164.6	151.4	4	9.0		
"	1750F-1/2HR-WQ+1000F-4HR-AC	162.7	150.6	4	6.0		
P41	" 2HR " " 8HR "	179.1	174.3	0	0.6		
"	" 16HR " " 24HR "	176.9	175.9	0	0.9		
"	As Cast	152.9	145.7	8	17.0		
"	1650F-2HR-WQ+1000F-24HR-AC	177.3	171.3	1	4.4		
"	" 16HR " " 8HR "	182.9	173.6	1	3.7		
P42	As Cast	146.1	132.8	13	2.0		
"	1750F-2HR-WQ+1000F-24HR-AC	171.5	161.7	3	7.6		
"	" 16HR " " 8HR "	172.0	161.9	3	6.7		
"	1650F-2HR " " " "	168.3	158.6	2	8.7		
"	" 16HR " " 24HR "	166.4	159.5	2	4.2		
"	1500F-2HR-FCE Cool to 1000F-AC	144.5	135.8	7	13.6		
P43	" " " " " " "	152.7	152.3	7	13.6		
"	1600F-2HR-WQ+900F-2HR-AC	174.0	163.4	3	3.6		
P63	" " " " " 4HR "	171	---	0	1.0		
"	" " " " " 8HR "	178.5	---	0	1.4		
P49	As Cast	153.1	142	8.5	6.0		
"	1600F-2HR-WQ+1000F-2HR-AC	168	---	2	4.0		
P57	As Cast	144	128.8	10	19.0		
"	1600F-2HR-WQ+1000F-4HR-AC	159.7	---	4	5.7		
"	" " " " " 2HR "	157.6	149.4	5	8.1		
P71	As Cast	151.9	138.3	8	12		
"	1600F-2HR-WQ+1000F-4HR-AC	149.4	143.7	2	1.8		
P42	1500F-2HR-WQ+900F-2HR "	163.6	143.1	3	5.5		
P40	1550F-2HR-WQ+900F-2HR-AC	174.7	162.0	3	6.7		
"	" " " 1000F-2HR "	168.7	162.5	3	3.9		
P51	" " " " 4HR "	157.9	149.4	5	8.0		
P71	" " " 1100F-2HR "	165.6	162.8	5	9.5		
"	" " " " 4HR "	164.5	161.2	3	7.0		
47	As Cast	144	154.7	12	20		338
"	1100F-2HR-AC	150	158.0	6.0	13.5	206.8	331
"	1550F-2HR-WQ+1000F-4HR "	150	162.8	6	12.9		341
"	1100F-2HR-AC+1550F-2HR-WQ +1000F-4HR-AC	140	140.0	1	6.3		363
48	As Cast	142	154.3	12.5	16		338
"	1100F-2HR-AC	140	154.0	3	16.4	217.6	321
"	1550F-2HR-WQ+1000F-4HR "	151	160	3	7.6	233.2	341
"	1100F-2HR-AC+1550F-2HR-WQ +1000F-4HR-AC	152	164.8	4	10.6	226.8	352

TABLE J19 (Continued)

MECHANICAL PROPERTIES OF HEAT TREATED CAST Ti-6Al-4V

Heat No.	Heat Treatment	Yield Strength KSI (.2%)	Ultimate Tensile Strength KSI	Elongation In 4D Per Cent	Reduction In Area Per Cent	Notch Tensile KSI	Brinell Hardness Number
45	As Cast	131	145	11	22.0		
"	1100F-8HR-AC	137	152	9	25.9	223	
"	1550F-2HR-WQ						
"	+1000F-4HR-AC	143	156	6	12.2	220	
"	1100F-8HR-AC+1550F-2HR-WQ						
"	+1000F-4HR-AC	145	158	6	11.5	226	
47	As Cast	144	155	12	20		
"	1100F-8HR-AC	150	158	6	13.5	207	
"	1550F-2HR-WQ						
"	+1000F-4HR-AC	150	163	6	12.9		
"	1100F-8HR-AC+1550F-2HR-WQ						
"	+1000F-4HR-AC	148	149	1	6.3		
48	As Cast	142	154	12	18		
"	1100F-8HR-AC	140	154	9	16.4	218	
"	1550F-2HR-WQ						
"	+1000F-4HR-AC	151	160	3	7.6	233	
"	1100F-8HR-AC+1550F-2HR-WQ						
"	+1000F-4HR-AC	152	165	4	10.6	227	
50	As Cast	135	150	12	18.0		
"	1100F-8HR-AC	132	147	7	18.9	216.8	
"	1550F-2HR-WQ						
"	+1000F-4HR-AC	137	152	5	12.6	218.0	
"	1100F-8HR-AC+1550F-2HR-WQ						
"	+1000F-4HR-AC	140	155	5	9.2	217.6	
70	As Cast	136	149	15	23		
"	1100F-8HR-AC	138	153	10	24.6	228	
"	1550F-2HR-WQ						
"	+1000F-4HR-AC	154	168	4	11.5	232	
"	1100F-8HR-AC+1550F-2HR-WQ						
"	+1000F-4HR-AC	158	172	3	8.6	228	
50	As Cast	135	150	12	18		324
"	1100F-8HR-AC	132	146.8	7	18.9	216.8	311
"	1550F-2HR-WQ+1000F-4HR-AC					218.0	341
"	1100F-8HR-AC+1550F-2HR-WQ						
"	+1000F-4HR-AC					217.6	341
70	As Cast	136	146.9	15	23		337
"	1100F-8HR-AC	138	153.2	10	24.6	228	321
"	1550F-2HR-WQ+1000F-4HR-AC	154	168	4	5	321.6	341
"	1100F-8HR-AC+1550F-2HR-WQ						
"	+1000F-4HR-AC	158	172	3	8.6	228	352
115	As Cast	132	143	8	13.2	211	
"	1150F-2HR-AC	136	142	9	13.6	212	
191	As Cast	125	134	8	8.8	203	
"	1150F-4HR-AC	127	133	9	20.3	205	
194	As Cast	126	136	6	13.8	201	
"	1100F-2HR-AC	127	137	8	14.1	195	
201	As Cast	124	135	7	10.2	203	
"	1100F-4HR-AC	126	136	7	10	196	
206	As Cast	123	134	7	13.5	201	
"	1000F-4HR-AC	123	133	7	15.1	206	
209	As Cast	129	137	7	16.2	204	
"	1000F-2HR-AC	131	141	7	20.7	213	
212	As Cast	133	141	7	16.3	204	
"	1200F-2HR-AC	135	141	9	4.4	204	
304	As Cast	134	144	8	11.2	209	
"	1200F-4HR-AC	136	143	5	6.3	211	
307	As Cast	135	144	6	13.9	206	
"	1200F-2HR-AC	133	144	5	11	207	

TABLE J19 (Continued)

MECHANICAL PROPERTIES OF HEAT TREATED CAST Ti-6Al-4V

Heat No.	Heat Treatment	Yield Strength KSI (.2%)	Ultimate Tensile Strength KSI	Elongation In 4D Per Cent	Reduction In Area Per Cent	Notch Tensile KSI	Brinell Hardness Number
308	As Cast	134	144	7	9.8	211	
"	1150F-4HR-AC	136	145	8	12.1	217	
309	As Cast	134	145	6	6.5	205	
"	1150F-4HR-AC	135	145	8	12.1	214	
P211-1	As Cast	142	124	9	15.1		305
"	1550F-2HR-WQ+1000F-4HR-AC	136	146.8	6	11.2		321
"	" " " 1100F-2HR "	136	145.6	5	10.5		321
"	" " " 1100F-2HR "	134	144.0	7	9.7		321
"	" " " 1000F-4HR-AC	135	146.0	7	12.9		321
P245	1600F-2HR-WQ+1000F-2HR-AC	119	138.2	7	17.5		302
"	" " " " 4HR "	119	138.2	6	14.5		302
"	" " " " 1100F-2HR "	117	132.2	5	10.5		321
"	" " " " 4HR "	116	133.2	7	14.2		311
"	" " " " 1000F-2HR "	120	156.0	5	10.0		302
"	" " " " 4HR "	119	135.6	3	8.6		302
"	" " " " 1100F-2HR "	116	130.8	4	13.5		311
"	" " " " 4HR "	116	131.6	6	10.5		293
356	1550F-2HR-WQ+1000F-2HR-AC	134	144.8	4	11.2		321
"	" " " " 4HR "	135	144.0	3	5.2		331
"	" " " " 1100F-2HR "	132	143.2	4	10.0		311
"	" " " " 4HR "	132	141.6	5	18.1		321
45	As Cast	131	145.1	11.0	22.0		334
"	1100F-8HR-AC	137	151.6	9.0	25.9	222.8	293
"	1550F-2HR-WQ+1000F-4HR "					220.0	331
"	1100F-8HR-AC+1550F-2HR-WQ+1000F-4HR-AC					226.4	341

TABLE 110

MECHANICAL PROPERTIES OF AS CAST EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Yield Strength, KSI	Ultimate Strength, KSI	Elong. in 4D, Per Cent	Reduction in Area, Per Cent	Tensile, KSI	Brinell Hardness Number	Carbon, Per Cent	Oxygen, Per Cent	Nitrogen, Per Cent
5	Ti 6Al-4V	P51-2	137.15	152.8	12.5	18.5			.084	.181	.028
5	Ti 6Al-4V	P54-2	143.2	152.9	9.5	14.0			.080	.200	.031
5	Ti 6Al-4V	F73-3	143.2	155.4	12.0	18.0			.094	.19	.036
6	Ti5Al2 1/2Sn	P26-28	127.15	136.7	9.0	16.0			.067	.230	.014
			129.0	136.7	11.0	20.0					
			128.2	137.4	11.0	20.0					
6	Ti5Al2 1/2Sn	P26-2C									
		A					205.65				
		B					206.5				
		C					207.3				
6	Ti5Al2 1/2Sn	P33-2A							.070	.256	.015
		A	132.1	139.5	11.0	22.0					
		B	132.25	139.6	11.5	22.0					
		C	133.0	139.8	11.0	22.0					
6	Ti5Al2 1/2Sn	P33-2B									
		A					203.3				
		B					196.45				
		C					204.2				
9	Ti 6Al-4V	P47-2	143.5	154.7	12.0	20.0			.087	.240	.027
9	Ti 6Al-4V	P50-2	134.95	150.0	12.0	18.0			.054	.158	.022
10	Ti 13V-11Cr	P62-2A	120.5	122.4	16.0	27.0			.028	.146	.027
		B	120.75	123.8	16.0	25.0					
		C	121.0	123.9	16.0	26.0					
10	Ti13V-11Cr	P62-3A					174.9				
		B					182.9				
		C					192.1				
10	Ti13V-11Cr	P66-2A	124.6	125.5	9.5	16.0			.023	.19	.028
		B	124.5	123.5	6.5	13.0					
		C	124.6	125.1	6.0	9.0					
10	Ti13V-11Cr	P66-3					174.7				
							180.7				
							178.5				
11	Ti13V-11Cr-1 1/2Al	P55-1A	125.9	126.75	13.5	23.0			.028	.153	.022
		B	125.25	126.5	18.5	24.0					
		C	125.25	126.1	15.0	23.0					
11	Ti13V-11Cr-1 1/2Al	P55-2A					187.3				
		B					184.8				
		C					174.7				
11	Ti13V-11Cr-1 1/2Al	P61-1A	126.4	127.8	10.5	17.0			.032	.1575	.027
		B	128.4	129.6	12.5	21.0					
		C	127.7	131.0	12.0	18.0					
11	Ti13V-11Cr-1 1/2Al	P61-2A					163.1				
		B					175.5				
		C					179.0				
11	Ti13V-11Cr-1 1/2Al	435	120	130.8	7.0	11.5	179.2	277	.031	.12	.022
			119	129.6	7.0	9.7					
12	Ti13V-11Cr	P33-2A	123.9	124.5	11.5	22.0			.020	.1438	.017
		B	121.95	124.7	9.5	22.0					
		C	121.06	128.4	10.0	23.0					

TABLE J20 (Continued)

MECHANICAL PROPERTIES OF AS CAST EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Yield Strength, KSI	Ultimate Strength, KSI	Elong. in 4D, Per Cent	Reduction in Area, Per Cent	Torch Tensile, KSI	Brinell Hardness Number	Carbon, Per Cent	Oxygen, Per Cent	Nitrogen, Per Cent
12	Ti13V-11Cr-2 1/2Al	P38-2B					191.1 176.8 180.65				
12		P44-2A	128.55	130.7	10.5	20.0			.030	.143	.020
		B	129.45	133.1	8.5	18.0					
		C	129.8	131.4	3.5	9.0					
12		P44-3A					165.6 170.2 162.25				
		B									
		C									
13	Ti13V-11Cr-4Al	P46-2A	131.4	133.1	9.0	21.0			.019	.143	.018
		B	134.65	136.15	9.0	21.0					
		C	133.6	135.8	8.5	18.0					
13		P46-3A					177.6 178.1 169.3				
		B									
		C									
13		P52-1A	133.3	133.1	9.0	21.0			.024	.153	.018
		B	135.2	136.15	9.0	21.0					
		C	134.3	135.8	8.5	18.0					
13		P52-2A					185.7 174.3 173.7				
		B									
		C									
14	8V-5Fe	P139-2A	None	182.4	0	0	102.8	438	.022	.101	
		B	None	180.0	0	0	104.0				
		C	None	186.0	0	0	126.0				
15	8V-5Fe-1Al	P140-2A	169.0	174.0	4	11.9	203.0	363	.030	.096	.014
		B	169.0	174.0	4	10.0	206.8				
		C	162.0	171.6	5	16.4	170.0				
15	8V-5Fe-1Al	358	150	178.8	2	4.0	178.4	355.2	.028	.11	.015
			150	180.0	2	5.3					
			150	180.0	2	5.0	178.0				
		140	167	174.0	4	13.0	193.0		.030	.096	.014
		521	---	181.0	2.0	4.9			.017	.060	.011
16	5 1/2Al-31/2V 0.45Fe-25Sn- 0.25Cu	P65-2A	125.4	142.0	10.5	15.0	210.0	311	.019	.10	.010
		B	126.7	142.4	10.0	16.0	206.8				
		C	125.9	142.0	10.0	15.0					
		P65-3A					217.5 216.0 215.1				
		B									
		C									
17		P151-2A	96.0	106.0	12	31.9	172.8	262	.024	.11	.0061
		B	96.0	106.2	10	36.5	173.2				
		C	98.0	108.2	11	29.2	172.0				
19	Ti7Al-3Mo	P64-2	112.3	128.4	8.0	18.0			.020	.131	.0083
			118.0	130.7	5.0	12.0					
			113.05	128.4	7.0	16.0					
19		P64-3					190.7 189.7 186.95				
19		P68-2	117.0	132.55	5.0	15.0			.024	.10	.010
			115.8	129.95	3.0	7.0					
			112.1	127.3	7.0	15.0					

TABLE J20 (Continued)

MECHANICAL PROPERTIES OF AS CAST EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Yield Strength, KSI	Ultimate Strength, KSI	Elong. In 4D, Per Cent	Reduction In Area, Per Cent	Notch Tensile, KSI	Brinell Hardness Number	Carbon, Per Cent	Oxygen, Per Cent	Nitrogen, Per Cent
20		P150-2A	128.0	136.0	10	25.7	190.0	277	.036	.13	.018
		B	128.2	137.2	10	24.6	204.2				
		C	128.0	136.0	11	27.2	190.0				
21		P152-1A	104.0	124.2	9	26.5	196.0	277	.036	.11	.0076
		B	105.0	126.0	9	18.1	194.0				
		C	106.7	127.6	10	25.2	190.0				
24	7Al-4Mo	P148-2A	112.0	132.0	5	15.1	194.0	305	.022	.072	.0046
		B	112.0	130.0	5	12.2	193.2				
		C	110.0	128.0	7	12.9	195.6				
25		P165-1A	90.0	114.0	10	18.9	172.0	250	.029	.089	.0069
		B	91.0	114.0	11	23.9	173.6				
		C	90.0	114.0	12	25.2	176.0				
26		P166-1A	107.0	121.0	9	25.2	190.0	287.6	.022	.11	.0067
		B	107.0	119.6	9	26.5	186.0				
		C	106.6	120.8	10	26.5	190.8				
27	4Al-3Mo-1V	P147-2A	91.0	110.0	15	37.0	172.0	254	.032	.086	.0056
		B	92.0	110.2	15	36.0	173.2				
		C	94.0	111.6	9	26.5	172.4				
28		P180-1A	134.0	140.8	1	5.3	220.8	321	.0035	.10	.012
		B	128.0	144.0	4	4.5	206.8				
		C	130.0	149.6	5	4.5	212.8				
29	6 1/2Al-3 1/2Mo-1V	P149-2A	136.0	154.8	8	14.0	216.4	331	.073	.020	.018
		B	132.0	152.0	12	21.5	224.0				
		C	135.0	154.2	10	20.4	214.4				
30		359	112	150.0	9	16.7	219.2	321	.062	.19	.018
			110	148.8	6	10.6					
		P189-1A	88.0	96.0	4	17.8	160.0	223	.044	.077	.0053
31		B	91.0	98.0	11	45.5	157.2				
		C	90.0	97.2	10	37.7	158.0				
		P190-1A	90.0	96.4	10	40.7	168.0	241	.024	.08	.0038
32		B	92.0	96.4	13	34.4	166.0				
		C	90.0	98.0	12	42.0	172.0				
		P146-2A	115.0	128.8	3	9.2	122.8	293	.032	.081	.018
33		B	115.0	134.0	5	11.5	167.2				
		C	115.0	135.2	6	18.6	177.6				
		P181-1A	149.0	165.2	1	0.6	170.0	345	.027	.115	.0041
34		B	145.0	158.2	1	0.6	168.0				
		C	146.0	161.2	1	1.6	172.0				
		P182-1A	129.0	148.0	2	3.8	173.2	317.0	.026	.084	.0051
35		B	134.0	141.2	2	1.6	180.4				
		P205-3A	None	106.2	1	0	125.6	331	.033	.084	.0085
		B	None	108.2	1	0	154.8				
36		C	None	118.0	1	0	122.0				
		P204-2A	Too ductile to draw curve	82.8	18	34.3	122.0	190	.036	.085	.0083
		B		83.2	18	33.7	124.0				
		C		83.2	17	35.8	122.0				

TABLE J20 (Continued)

MECHANICAL PROPERTIES OF AS CAST EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Yield Strength, KSI	Ultimate Strength, KSI	Elong. In 4D, Per Cent	Reduction In Area, Per Cent	Notch Tensile, KSI	Brinell Hardness Number	Carbon, Per Cent	Oxygen, Per Cent	Nitrogen, Per Cent
37		P203-2A	None	136.8	1	1.6	122.8	375	.029	.105	.0070
		B	None	138.2	1	.7	100.8				
		C	None	138.0	0	2.4	116.4				
38		P193-1A	Too ductile	96.4	12	27.2	148.0				
		B		92.0	18	27.2	148.0				
		C		95.6	10	19.7	140.0				
39	5Mo-6V-2Ni	P164-1A	138.0	154.0	5	13.5	204.8	314.3	.021	.095	.011
		B	137.0	152.0	4	11.0	204.0				
		C	140.0	156.2	4	10.6	204.8				
40		409	140	156.0	8	13.5	217.6	314	.027	.108	.0092
			140	156.0	7	14.5					
		P233-1A	146.0	160.0	7	13.4	226.0	331	.026	.064	.0095
41	2Cu	B	146.0	160.0	7	10.5	232.0				
		C	145.0	159.2	5	10.6	228.0				
		P202-2A	62.0	79.6	25	38.3	112.0	176	.038	.10	.032
		B	62.0	80.0	24	36.5	114.0				
		C	61.0	79.6	25	39.3	114.0				
		P202	62.0	80.0	25	38.0	114.0				
		P523	85.0	106.0	15	25.9	156.0				
		P523	88.0	104.0	16	22.4		235	.033	.20	.035
		P523	89.0	106.0	15	24.6		248			
		P523	88.0	106.0	15	23.9		241			
		P579	100.0	117.0	14	18.9	173.0	255	.045	.27	.047
		P579	100.0	118.0	13	19.7	174	255			
		P579	100.0	117	13	18.1		262			
		P579	102.0	118.0	13	17.5		262			
		P579	98.0	115.0	6.0	9.2		269			
		P579	102.0	118.0	13	22.4		248			
42		P223-1A	111.0	123.6	8	15.8	182.4	283	.025	.08	.0041
		B	110.0	122.2	9	18.9	182.8				
		C	111.0	122.4	9	15.8	182.8				
43		P267-1A	103.0	123.8	9	27.2		272	.024	.066	.0072
		B	103.0	122.8	10	21.7					
		C	102.0	122.2	12	34.4					
44	3Al-2 1/2V	P144-1A	94.0	106.8	11	28.5	168.4	253	.028	.12	.019
		B	94.0	108.0	11	29.8					
		C	94.0	108.0	12	29.9					
45		P222-1A	112.0	124.4	12	30.5	194.4	288	.019	.087	.0047
		B	113.0	125.6	14	27.2	192.0				
		C	114.0	126.4	16	25.9	196.8				
46		P182-2A	68.0	92.0	14	25.2	127.2	201	.021	.12	.012
		B	68.0	90.4	13	26.0	132.0				
		C	63.0	91.6	13	23.0	133.2				
47		P226-1A	122.0	136.8	11	27.9	206.0	290	.030	.079	.012
		B	122.0	136.8	12	24.6	206.0				
		C	121.0	136.0	13	27.9	208.6				
48		P183-2A	137.0	156.0	4	4.9	186.4		.018	.16	.013
		B	136.0	154.0	4	6.1	180.0				
		C	136.0	154.4	3	6.3	178.0				

TABLE J20 (Continued)

MECHANICAL PROPERTIES OF AS CAST EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Yield Strength, KSI	Ultimate Strength, KSI	Elong. in 4D, Per Cent	Reduction in Area, Per Cent	Notch Tensile, KSI	Brinell Hardness Number	Carbon, Per Cent	Oxygen, Per Cent	Nitrogen, Per Cent
49		P234-1A	111.0	132.0	10	32.5	206.0	296	.019	.073	.024
		B	115.0	136.0	8	21.0	204.0				
		C	111.0	131.0	9	29.0	204.4				
50		P230-1A	122.0	142.0	4	5.3	196.2	305	.012	.19	.0054
		B	123.0	143.2	11	18.9	186.0				
		C	123.0	144.0	10	18.1	190.0				
51		P236-1A	None	102.8	0	0	90.0	331	.022	.15	.0054
		B	None	100.0	0	0	76.0				
		C	None	106.4	0	0	88.0				
52			Not testable		Too brittle						
53	6Al-4V-0.15S	P267-1A	114.0	133.6	8	17.5	44.8	280	.036	.12	.0078
		B	112.0	132.0	10	21.0	49.2				
		C	112.0	131.6	12	33.0	54.4				
54	6Al-4V-0.5S	P273-1A	None	76.0	0	0	62.0	282	.028		.0077
		B	None	80.0	0	0	63.2				
		C	None	90.4	0	0	78.0				
55	6Al-4V-0.5W	P297-1A	104.0	124.0	9	25.7		277	.025	.078	.0067
		B	106.0	126.0	10	25.2					
		C	104.0	124.0	12	41.4					
56	6Al-4V-0.5Ta	348	104.0	124.0	11	38.9	200	243	.031	.12	.015
			108.0	128.0	11	24.9	197.6				
			106.0	124.0	11	33.6	203.2				
57	3Al-7Mo-0.258e	P298-1A	114.0	134.0	6	10.6	192.0	285	.016	.106	.0047
		B	113.0	134.0	8	12.2	189.6				
		C	112.0	133.2	7	10.6	192.0				
59	4Al-4Sn-8Zr	P299-1A	112.0	123.6	10	23		262	.017	.10	.0045
		B	108.0	120.0	10	23.7					
		C	110.0	124.0	10	24.4					
60	4Al-4Sn-8Zr-1Fe-1Cr-1V	P300-1A	132.0	151.6	9	21.5	202.0	293	.034	.11	.020
			132.0	152.0	10	22.4	204.8				
			136.0	154.0	10	19.7	205.6				
61	7Al-2Mo-3Cr-35Sn-2Zr	P489	151.0	170.0	7	10.5		352	.048	.29	.063
		P489	156.0	177.0	4	7.0					
		P489	156.0	156.0	0	0	161.0				
62	7Al-2Mo-3Cr-35Sn-2Zr	P301-1A	156.0	168.2	1	1.6	130.0	371	.014	.12	.064
		B	156.0	172.8	2	4.0	133.2				
		C	154.0	164.0	1	0.7	137.2				
63	7Al-2Mo-0.5W-0.5Ta-3Cr-35Sn-2Zr	349	151.0	161.2			160.8	359.6	.0030	.12	.0082
			154.0	174.0	3	7.0	156.2				
			153.0	172.0	1	3.3	166.8				
64	7Al-2Mo-0.5W-0.5Ta-3Cr-35Sn-2Zr	350	155.0	172.0	1	3.3	128.0	363	.022	.11	.0073
			155.0	162.4	1	1.6	128.0				
			156.0	170.8	1	4.0	152.0				

TABLE J20 (Continued)

MECHANICAL PROPERTIES OF AS CAST EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Yield Strength, KSI	Ultimate Strength, KSI	Elong. In 4D, Per Cent	Reduction In Area, Per Cent	Notch Tensile, KSI	Brinell Hardness Number	Carbon, Per Cent	Oxygen, Per Cent	Nitrogen, Per Cent
64	4Al-4Sn-8Zr-0.07N	P302-A B C	124.0 126.0 125.0	141.2 142.0 142.8	11 10 10	22.4 21.5 23.0	213.2 214.0 214.4	290	.010	.12	.068
65	4Al-2Mo-2Cr-2V-2Fe-4Sn-8Zr	351		183.2 137.6 190.8	1 1 0	0 0 0	150.8 131.6 130.0	405.2	.018	.15	.0099
66	2Al-5Fe-2Sn-8V-5Zr	352	131 131 130	140.0 140.0 130.0	2 22 2	3.0 0.7 1.6	154.8 116.0 118.0	285.1	.023	.18	.015
68	2Al-8V-5Fe	354	140 142 143	154.4 153.2 153.6	7 4 4	11.9 8.6 10.6	183.6 186.8 148.0	311	.026	.11	.014
69	6Al-4V-1Cr-1Mo	434 434 434 None-A B	129 126 128 130 130	147.6 144.8 145.0 152.0 152.0	9 7 8 7 6	15.1 18.6 17.0 15.1 9.2	204 204 204 204 204	317 317 317 321 321	.038	.12	.0063
71	6Al-4V-0.048C-0.06Ox-0.028N	355	113 111 114	128.0 127.2 128.0	7 6 6	11.5 12.2 12.9	196.0 190.4 190.8	277	.048	.06	.028
72	6Al-4V-0.028C-0.050Ox-0.045N	357	114 115 115	130.8 133.2 132.0	6 7 7	15.1 15.1 15.1	200.0 196.8 198.0	287.6	.028	.05	.045
73	6Al-4V-0.025C-0.06Ox-0.079N	356	122 124 121	136.8 142.8 138.8	8 6 8	15.1 11.2 16.4	200.4 212.0 206.0	290.3	.025	.06	.079
80	0.04C-0.18Ox-0.018N	341	64 64 63	82.8 80.2 81.8	15 15 13	25.9 23.0 34.4			.040	.18	.018
84	.036C-0.28Ox-0.018N	431	48	90.8	13	21.7			.036	.28	.018
84	.053C-.26Ox-.021N	432	76	96.0	13	28.5	213.0	213	.053	.26	.021
90	.03O ₂ , 15Al, 1V		Not castable - too brittle								
96		513	54	71.0	20	40.0	108		.025	.099	.016
97		514	69	84.0	20	38.7	122		.037	.077	.073
98		520	111	127.0	3	0.7	119		.053	.57	.016
99		519	117	121.0	1	1.6	90		.036	.63	.066
100		518	59	83.0	16	27.0	128		.188	.13	.013
101		517	71	94.0	11	16.7	138		.190	.14	.059
102		516		85.0	1	0.8	86		.183	.67	.013
103		515		93.0	1	2.1	68		.185	.61	.076
104		P142	135	157.0	8	17.6			.100	.196	.035
105	4Al-4Sn-8Zr-1.5Fe-1.5Cr-1.5V	P596 P596 P596	154 154 154	171.0 171.0 170.0	4 5 4	5.3 9.7 5.3	216 216 219	352 352 352	.060	.12	.038
106	4Al-4Sn-8Zr-.5Fe-.5Cr-.5V	P597 P597 P510 P511 P526 P531 P536 P541 P548 P554 P559	132 132 144 131 135 130 132 132 134 129 131	146.0 149.0 158.0 146.0 153.0 148.0 152.0 148.0 150.0 146.0 149.0	10 11 4 8 7 7 8 8 7 8 9	22.4 16.7 5.1 12.9 13.5 19.4 10.6 13.4 13.4 14.5 17.8	219 219 219 219 219 219 219 219 219 219 219	321 321 321 321 321 321 321 321 321 321 321	.060	.11	.038
									.055	.127	.153
									.050	.151	.011
									.055	.21	.025
									.050	.21	.020
									.050	.22	.024
									.057	.23	.031
									.067	.22	.055
									.048	.23	.028
									.026	.23	.052

TABLE J20 (Continued)

MECHANICAL PROPERTIES OF AS CAST EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Yield Strength, KSI	Ultimate Strength, KSI	Elong. In 4D, Per Cent	Reduction In Area, Per Cent	Notch Tensile, KSI	Brinell Hardness Number	Carbon, Per Cent	Oxygen, Per Cent	Nitrogen, Per Cent
107	10Cr-1Al	606		140.0	0	0					
				160.0	1	1.6					
				172.0	1	3.3					
108	8Zr-4Al-4Sn-2V	607	130	152.4	11	21.0	208		.077	.030	.035
110	10Zr-4Al-4Sn-2V	626	130	152.8	11	20.5					
			118	135.2	11	27.9	201		.016	.11	.014
111	8Zr-4Al-6Sn-2V	627	117	134.2	11	24.6					
			118	140.0	12	22.4	198		.016	.12	.014
112	8Zr-6Al-4Sn-2V	628	118	141.0	10	22.4					
			120	145.0	12	21.4			.021	.12	.013
			120	144.0	11	23.2					

TABLE J21

MECHANICAL PROPERTIES OF HEAT TREATED EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Heat Treatment	Yield Strength ksi (0.2% Offset)	Ultimate Tensile Strength ksi	Elongation in 40 Per Cent	Reduction in Area Per Cent	Batch for the KSI	Brinell Hardness Number
6	5Al-2 $\frac{1}{2}$ Sn	P20-20	As-Cast	122.15	146.7	9.0	16.0		
				121.0	146.7	11.0	20.0		
				120.2	147.4	11.0	20.0		
		P20-20	As-Cast						
		A							205.65
		B							206.5
		C							207.3
		P33-20	As-Cast						
		A		112.1	139.5	11.0	22.0		
		B		112.25	139.5	11.5	22.0		
		C		113.0	139.5	11.0	22.0		
		P33-20	As-Cast						
		A							203.3
		B							196.45
		C							204.2
10	13V-11Cr -0Al	P62	As-Cast	116.157	146.555	13.0	24.7		
			1700F-2HR-WQ 925F-4HR-AL	142.97	150.13	4.0	5.6		
		"	" " " " " " " "	124.0	124.09	12.0	20.2		
		"	1700F-2HR-WQ " " " "	152.91	161.54	6.0	3.2		
		"	" " " " " " " "	122.53	127.53	12.0	33.2		
		"	" " " " " " " "	126.7	127.1	15.0	33.5		
		"	" " " " 925F-2HR-AL	126.6	132.5	6.0	10.7		
		"	" " " " 1050F " " "	122.9	126.2	6.0	7.4		
		"	" " " " 925F-4HR " "	132.4	145.0	5.0	6.1		
		"	" " " " 1050F " " "	122.4	131.0	11.0	13.0		
		"	" " " " 925F-4HR-AL	151.29	151.35	8	11.1		
		P65	1700F-2HR-WQ	130.6	131.0	11.0	14.7		
		"	" " " " 925F-1HR-AL	130.9	132.0	6.0	6.7		
		"	" " " " 1050F " " "	126.1	129.7	4.0	5.7		
		"	" " " " 925F-4HR-AL	139.0	147.6	4.0	4.9		
18	13V-11Cr -0Al	P66	1700F-2HR-WQ 1050F-4HR-AL	128.7	135.3	7.0	6.5		
		"	" " " " 925F-16HR-AL	150.3	161.5	4.0	3.8		
		"	" " " " 1050F " " "	122.8	135.0	5.0	7.2		
11	13V-11Cr 1 $\frac{1}{2}$ Al	P55-1A	As-Cast	125.9	126.75	13.5	23.0		
		B	As-Cast	125.25	126.5	18.5	24.0		
		C	As-Cast	25.25	26.1	15.0	23.0		
		-2A	As-Cast						167.3
		B	As-Cast						164.8
		C	As-Cast						174.7
		P61-1A	As-Cast	126.4	127.8	13.5	17.0		
		B	As-Cast	126.9	129.6	12.5	21.0		
		C	As-Cast	127.7	131.0	12.0	16.0		
		P61-2A	As-Cast						163.1
		B	As-Cast						175.5
		C	As-Cast						179.0
		P55	1700F-2HR-WQ	125.0	125.5	12.0	36.1		
		P61	" " " 925F-1HR-AL	133.6	135.7	5.0	11.0		
		"	" " " 1050F " " "	130.5	136.2	10.0	12.0		
		P55	" " " 925F-2HR " "	126.8	131.6	11.0	16.9		
		"	" " " 1050F " " "	126.3	126.5	6.0	13.3		
		P61	" " " 925F-4HR " "	142.4	156.0	4.0	6.2		
		"	" " " 1050F " " "	146.0	163.3	6.0	10.3		
		P61	" " " 925F-4HR " "	129.5	140.1	11.0	12.9		
		"	" " " 1050F " " "	133.5	140.6	9.0	13.6		
		P61	" " " 925F-1HR " "	174.9	133.6	5	6.6		
		"	" " " 1050F " " "	140.0	147.1	6	4.3		
		"	" " " " " " " "	133.3	133.3	12	23.4		
		P55	As-Cast	125.25	125.4	14	20.9		
		"	1700F-2HR-WQ 925F-1HR-AL	140.20	142.5	3	5.1		
		"	" " " " " " " "	125.20	125.525	14	20.5		
		"	1700F-2HR-WQ 925F-4HR-AL	140.20	140.20	4	4.0		
		"	" " " " " " " "	125.2	125.30	4	10.3		
		P55	" " " " " " " "	125.2	125.3	4	6.1		
		"	" " " " " " " "	125.2	125.3	4	6.1		

TABLE J21 (Continued)

MECHANICAL PROPERTIES OF HEAT TREATED EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Heat Treatment	Yield Strength KSI (0.2% Offset)	Ultimate Tensile Strength KSI	Elongation In 4D Per Cent	Reduction In Area Per Cent	Notch Tensile KSI	Brinell Hardness Number
11	13V-11Cr 1 1/2Al	P55	1400F-2HR-WQ+925F-48HR-AC	160	176	3	5.1		
		"	" " " "	125	125	14	20.5		
		"	1700F- " " 925F-48HR-AC	170	180	4	4.0		
		"	" " " "	127	127		30.3		
		"	" " " "	126	126	17	36.1		
		P61	" " " 14925F-1HR-AC	135	136	5	11.0		
		"	" " " 1050 " "	130	136	10	12.0		
		P55	" " " 925F-2HR-AC	129	132	11	18.9		
		"	" " " 1050F " "	125	126	6	13.3		
		P61	" " " 925 1HR- " "	147	156	4	8.2		
		"	" " " 1050 " " "	148	163	6	10.3		
		P55	" " " 925 8HR " "	140	149	11	12.9		
		"	" " " 1050 " " "	134	141	9	13.8		
		P61	" " " 925 16HR " "	180	194	5	6.8		
		"	" " " 1050 16 " "	140	147	6	4.8		
		"	" " " "	133	133	12	23.4		
		P55	As Cast	125	125	14	20.9		
		"	1100F-2HR-WQ-925F-12HR-AC	159	174	6	9.2		
		"	1600F-2HR-WQ+ " " " "	157	174	6	10.6		
		"	1700F " " 1000F-8HR- " "	143	160	8	12.9		
		"q	1600 " " " " " "	143	158	8	12.7		
		P61	1600 " " 950F- " " "	150	166	5	6.7		
		"	" " " " 16HR " "	153	170	5	6.7		
		P55	" " " " 24HR " "	153	170	5	7.6		
		"	" " " 900F-8HR " "	140	154	10	15.1		
		"	" " " 16HR " "	152	169	7	10.0		
		"	" " " 24HR " "	158	175	5	6.3		
		435	As Cast	120	130.8	7.0	11.5	179.2	277
		"	As Cast	119	129.6	7.0	9.7		
		"	1550F-2HR-WQ+950F-8HR-AC	152	163	5	7.6		
		"	" " " 16HR " "	159	175	4	6.3		
		"	" " " 24HR " "	159	174	4	5.2		
		"	" " " 900 8HR " "	169	182	3	5.3		
		"	" " " 16HR " "	159	171	5	7.6		
		"	" " " 24HR " "	171	187	4	6.3		
		"	1500F " " 950F-8HR " "	149	166	5	5.3		
		435	1500F-2HR-WQ-950F-16HR-AC	156	174	4	6.3		
		"	" " " 24HR- " "	158	176	4	5.3		
		"	" " " 900F-8HR " "	151	165	5	6.1		
		"	" " " 16HR " "	165	182	3	6.3		
		"	" " " 24HR " "	166	183	4	5.2		
		"	1600F-2HR-WQ+1050F-2HR-AC						
		55	1700F-2HR-WQ+1050F-4HR-AC	133	146.0	7	13.5		311
		"	" " " 925F-12HR-AC	159	174.4	5.5	9.2		341
		"	1600F " " 1050F-4 HR- " "	128	142.0	7.5	11.9		302
		"	" " " 925F-12HR " "	157	174.4	6	10.6		341
		"	1700F " " 1000F-8HR " "	143	159.6	8	12.9		341
		"	1600F " " " " " "	143	158.0	8	12.7		311
12	13V-11Cr 2 1/2Al	P38	As Cast	125, 144	125, 172	11	19.1		
		"	1400F-2HR-WQ+925F-48HR-AC	172, 371	174, 433	1	0.0		
		"	" " " "	176, 340	176, 403	12	20.6		
		"	1700F " " 925F-48HR-AC	180, 922	194, 475	2	2.7		
		"	" " " "	127, 108	127, 316	12	25.0		
		"	" " " 925F-48HR-AC	164, 351	173, 437	2	2.6		
		"-7A	As Cast	121.9	124.5	11.5	22.0		
		" B	As Cast	121.95	124.7	9.5	22.0		
		" C	As Cast	121.0	125.4	10.0	23.0		
		"-2B	As Cast					191.1	
		"	As Cast					176.8	
		"	As Cast					169.55	
		P44-2A	As Cast	172.55	139.7	10.5	20.6		
		" B	As Cast	172.66	131.1	8.5	18.0		
		" C	As Cast	172.8	131.4	3.5	9.0		
		P44-3A	As Cast					165.6	
		" B	As Cast					170.2	
		" C	As Cast					162.25	

TABLE III (Continued)

MECHANICAL PROPERTIES OF HEAT TREATED EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Heat Treatment	Yield Strength KSI (0.2% Offset)	Ultimate Tensile Strength KSI	Elongation In 40 Per Cent	Reduction In Area Per Cent	Notch Tensile KSI	Brinell Hardness Number
		P44	1700F-2HR-WQ	133.7	133.7	18	25.6		
		"	" " " 925F-1HR-AC	135.6	135.9	4	6.2		
		"	" " " 1050F " "	137.5	139.6	6	9.2		
		"	" " " 925F-4HR "	138.3	142.6	7	10.6		
		"	" " " 1050F " "	135.6	139.8	9	10.7		
		"	" " " 925F-16HR-AC	176.1	189.0	4	4.6		
		"	" " " 1050F " "	143.4	148.9	4	6.5		
		P38	" " " "	125.7	126.7	12	29.6		
		"	" " " 925F-2HR-AC	141.4	144.0	4	2.8		
		"	" " " 1050F " "	147.2	150.2	9	14.0		
		"	" " " 925F-4HR-AC	165.5	176.5	6	9.5		
		"	" " " 1050F-8HR " "	140.6	144.7	8	13.5		
13	13V-11Cr-4Al	P46-2A	As Cast	131.4	133.1	9.0	21.0		
		B	As Cast	134.65	136.15	9.0	21.0		
		C	As Cast	133.0	135.8	8.5	18.0		
		-3A	As Cast					177.6	
		B	As Cast					178.1	
		C	As Cast					169.3	
		P52-1A	As Cast	133.3	133.1	9.0	21.0		
		B	As Cast	135.2	136.15	9.0	21.0		
		C	As Cast	134.5	135.8	8.5	18.0		
		-2A	As Cast					185.7	
		B	As Cast					174.3	
		C	As Cast					173.7	
		P46	As Cast	132.3	134.3	9	18.3		
		"	1400F-2HR-WQ+925F-4HR-AC		182.8	0	0.2		
		"	" " " "	131.2	133.7	7	14.7		
		"	1700F-2HR-WQ+ " " "	205.8	206.6	0	2.0		
		"	" " " "	132.018	133.956	15	31.1		
		P52	" " " "	136.7	137.3	13	26.6		
		"	" " " 925F-1HR-AC	140.8	162.8	4	8.2		
		"	" " " 1050F " "	146.6	148.8	7	11.2		
		"	" " " 925F-4HR " "	157.0	163.6	3	3.9		
		"	" " " 1050F " "	160.8	170.6	4	6.5		
		"	" " " 925F-16HR " "	192.6	198.7	2	1.4		
		"	" " " 1050F " "	175.5	179.5	3	2.0		
		P46	" " " "	132.5	133.8	13	32.3		
		"	" " " 925F-2HR-AC	133.7	136.8	3	2.7		
		"	" " " 1050F " "	146.6	148.4	5	8.6		
		"	" " " 925F-8HR-AC	169.0	176.3	3	4.1		
		"	" " " 1050F " "	163.6	171.7	3	2.9		
		"	" " " 925F-49HR AC		151.1	0	1.2		
14	8V-5Fe	P139-2A	As Cast	None	182.4	0	0	(P139-3)	
		B	As Cast	None	180.0	0	0	102.8	438
		C	As Cast	None	186.0	0	0	104.0	
		P139	1350F-2HR-WQ+1000F-1HR-AC	165.6	166.6	1	3.5	126.0	
		"	" " " " 2HR "	161.1	163.0	1	1.4		
		"	" " " " 4HR-AC	Not Taken	155.3	2	4.4		
		"	" " " " 1HR "	"	146.8	5	6.6		
		"	" " " " 2HR "	"	144.9	4	5.7		
		"	" " " " 4HR "	"	144.1	4	6.3		
		"	" " " " 8HR "	"	140.2	3	9.5		
		"	As Cast	"	182.8	0	0		
15	8V-5Fe 1A1	P140-2A	As Cast	169.0	174.0	4	11.9	203.0	363
		B	As Cast	169.0	176.0	4	10.0	206.8	
		C	As Cast	162.0	171.6	5	16.4	170.0	
		F358	As Cast	150	178.8	2	4.0	178.4	355.2
		"	As Cast	150	180.0	2	5.3		
		"	As Cast	150	180.0	2	5.0	178.0	
		P140	As Cast	152	174.0	4	13.0	153.0	
		P52i	As Cast	XXX	181.0	2.0	4.9		
		P140	1350F-2HR-WQ+1000F-1HR-AC	Not Taken	175.4	1	3.3		
		"	" " " " 2HR "	"	174.8	1	0.8		
		"	" " " " 4HR "	"	172.2	2	2.2		
		"	1250F-2HR " " 1HR "	167.5	175.0	3	2.6		
		"	" " " " 2HR "	Not Taken	174.0	5	5.1		
		"	" " " " 4HR "	"	171.7	4	4.3		
		"	" " " " 8HR "	"	170.8	4	5.1		
		"	" " " " 16HR "	167	170	"	4.8	172	(b)
		"	" " " " 32HR "	"	170	"	4.9		331

TABLE J21 (Continued)

MECHANICAL PROPERTIES OF HEAT TREATED EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Heat Treatment	Yield Strength KSI (0.2% Offset)	Ultimate Tensile Strength KSI	Elongation In 40 Per Cent	Reduction In Area Per Cent	Notch Tensile KSI (P140-3)	Brinell Hardness Number
15		P358	1000F-48HR-AC	160	166	3	9.5		331
		P140	900F- 8HR "	---	201	0	0	192	338
		P358	" 24HR "	---	192	3	7.2		363
		"	" 48HR "	---	181	2	3.4	172	351
		P140	800F- 8HR "	---	191	2	0		401
		P358	" 24HR "	---	195	1	0		388
		"	" 48HR "	---	194	1	0	118	375
		P140	700F- 8HR "	---	148	0	0		415
		P358	" 24HR "	---	121	0.5	2.1		401
		"	" 48HR "	---	122	0	0	85	415
		P140	600F- 8HR "	---					
		P358	" 24HR "	---					
16	5.5Al 5.5V-25Sn .5Fe-.25Cu	P65	As Cast	126	142.1	10	15.3		
		"	1150F-16HR-AC	127.7	138.9	9	16.2		
		P69	As Cast	125.2	140.4	9	16.2		
		"	1100F- 2HR-AC	130.6	141.0	9	19.6		
		"	" 4HR "	129.4	140.6	6	13.7		
		"	" 8HR "	129	140.6	5	13.2		
		"	" 16HR "	130.8	140.8	9	13.0		
		"	1150F- 2HR "	127.5	136.6	8	14.1		
		"	" 4HR "	128.1	139.1	11	17.8		
		"	" 8HR "	128.9	140.7	8	14.8		
		"	1050F- 2HR "	129.2	141.8	7	8.8		
		"	" 4HR "	129.9	142.9	9	11.8		
		"	" 8HR "	130.5	142.3	6	12.7		
		P65-2A	As Cast	125.4	142.0	10.5	15.0	210.0	311
		B	As Cast	126.7	142.4	10.0	16.0	206.8	
		C	As Cast	125.9	142.0	10.0	15.0		
		P65-3A	As Cast					217.5	
		B	As Cast					216.0	
		C	As Cast					215.1	
		P65	As Cast	126.0	142.1	10	15.3		
		"	1650F- 2HR-WQ+1000F-6HR-AC	172.0	181.2	0	0.6		
		"	" " " " 16HR	171.8	176.5	0	0.3		
		"	" " " " 1100F-6HR "	161.2	172.0	0	1.0		
		"	" " " " 16HR "	158.3	168.9	2	5.0		
		"	" 16HR 1000F-6HR "	164.9	173.3	0	0.5		
		"	" " " " 16HR "	164.6	175.7	1	2.8		
		"	" " " " 1100F-6HR "	161.1	168.2	1	3.6		
		"	" " " " 16HR "	156.7	165.8	1	4.8		
		P69	As Cast	125.2	140.4	9	16.2		
19	87Al-14In	P64-2	As Cast	112.3	128.4	8.0	16.0		
		"	As Cast	118.0	130.9	5	12.0		
		"	As Cast	113.05	128.4	7	16.0		
		P64-3	As Cast					190.7	
		"	As Cast					189.7	
		"	As Cast					186.95	
		P68-2	As Cast	117.0	132.55	5	15.0		
		"	As Cast	115.8	129.55	3	7.0		
		"	As Cast	112.1	127.3	7	15.0		
		P64	As Cast	119.3	129.3	7	15.0		
		"	1600F- 2HR AL+1100F-16HR-AL	119.3	124.9	4	14.2		
		"	" " " " WQ+1100F-16HR-AL	117.9	126.8	6	12.0		
		"	" " " " WQ+1100F-16HR-AL	113.5	116.5	3	9.2		
		"	" " " " 1600F- 2HR-AL+1100F-16HR-AL	117.3	127.0	5	12.2		
		"	" " " " 1600F- 2HR-AL+1100F-16HR-AL	115.5	125.0	6	12.7		
		"	" " " " WQ	108.7	122.0	4	12.9		
		"	" " " " WQ	107.7	120.7	5	12.2		

TABLE J21 (Continued)

MECHANICAL PROPERTIES OF HEAT TREATED EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Heat Treatment	Yield Strength KSI (0.2% Offset)	Ultimate Tensile Strength KSI	Elongation in 4D Per Cent	Reduction in Area Per Cent	Notch Tensile KSI	Brinell Hardness Number
19		P68	As Cast	115.0	129.9	5	12.0		
		"	1700F-2HR-WQ+1000F-1HR-AC	130.6	142.0	3	7.6		
		"	" " " " 2HR "	133.0	148.4	3	2.7		
		"	" " " " 4HR "	138.4	148.0	2	10.7		
		"	" " " " 8HR "	132.9	148.6	2	1.8		
		"	" " " " 1100F-1HR "	131.9	142.1	1	0.6		
		"	" " " " 2HR "	138.6	146.6	2	0.8		
		P64	" " " " 4HR "	131.8	138.6	2	2.1		
		"	1600F " " 1000F-4HR "	125.5	137.3	3	8.7		
		"	" " " " 1100F-2HR "	127.1	138.1	3	6.9		
20	10V-13Cr 5Al	P150-2A	As Cast	128.0	136.0	10	25.7	(150-3)	277
		B	As Cast	128.2	137.2	10	24.6	190.0	
		C	As Cast	128.0	136.0	11	27.2	204.2	
		P150	As Cast	128	136	10	25.0	190.0	
		"	1600F-2HR-WQ+1000F-2HR-AC	151	163	3	7.0	193	
		"	" " " " 4HR "	179	182	1	2.1		
		"	" " " " 8HR "	---	173	0.5	0.7		
		"	" " " " 1050F-2HR "	167	167	1	0.0		
		"	" " " " 4HR "	---	158	1	1.5		
		"	" " " " 8HR "	---	173	1	0.0		
24	7Al-1Mo	P148-2A	As Cast	112.0	132.0	5	15.1	194.0	305
		B	As Cast	112.0	130.0	5	12.2	193.2	
		C	As Cast	110.0	128.0	7	12.9	195.6	
		P148	As Cast	111.3	130.0	6	13.4		
		"	1600F-2HR-WQ+1000F-4HR-AC	127.2	145.8	4	5.7		
		"	" " " " 1100F-2HR "	129.0	140.2	2	6.7		
		"	" " " " 4HR "	129.0	142.5	3	1.4		
		P166-1A	As Cast	107.0	121.0	9	25.2	190.0	
		B	As Cast	107.0	119.6	9	26.5	186.0	
		C	As Cast	106.6	120.8	10	26.5	190.8	
26	8Al-1Mo IV	166	1600F-2HR-WQ+1000F-1HR-AC	107	122.0	4	4.9	285	287.6
		"	" " " " 2HR "	103	119.6	9	24.1	285	
		"	" " " " 4HR "	107	116.4	8	25.9	293	
		"	" " " " 900F-2HR "	104	117.2	7	21.7	285	
		"	" " " " 4HR "	106	119.6	10	22.4	293	
		P147-2A	As Cast	91.0	110.0	15	37.0	(P147-3)	
		B	As Cast	92.0	110.2	15	36.0	172.0	
		C	As Cast	94.0	111.6	9	26.5	173.2	
		P147	As Cast	92.3	110.6	13	33.2	172.4	
		"	1550F-2HR-WQ+1000F-4HR-AC	107.5	123.6	3	2.4		
27	4Al-3Mo IV	"	" " " " 1100F-2HR "	103.4	118.6	4	8.5		254
		"	" " " " 4HR "	101.6	116.4	6	11.3		
		P149-2A	As Cast	136.0	154.8	8	14.0	(P149-3)	
		B	As Cast	132.0	152.0	12	21.5	216.4	
		C	As Cast	135.0	154.2	10	20.4	224.0	
		359	As Cast	112	150.0	9	16.7	214.4	
		P149	As Cast	110	148.8	6	10.6	219.2	
		"	1600F-2HR-WQ+1000F-1HR-AC	134.2	153.2	10	12.9		
		"	" " " " 2HR "	125.6	171.1	2	3.3		
		"	" " " " 4HR "	168.0	173.5	1	3.0		
29	6Al-3Mo IV	"	" " " " 8HR "	170.0	175.5	2	3.2		321
		"	" " " " 2HR "	---	166.1	2	0.2		
		P149	As Cast	134	154	10	10.9		
		"	1600F-2HR-WQ+1000F-1HR-AC	165	171	2	3.8	218	
		"	" " " " 2HR "	168	174	1	3.0		
		"	" " " " 4HR "	170	176	2	3.2		
		"	" " " " 8HR "	---	166	2	0.2		
		"	1550F " " " 4HR "	158	164	2	3.2		
		"	" " " " 2HR "	159	168	2	4.0		
		"	" " " " 1100F-2HR "	155	170	3	7.0		
29	6Al-3Mo IV	"	" " " " 4HR "	157	167	4	7.4		321
		"	1500F " " " 1000F " "	136	170	3	6.3		
		"	" " " " 8HR "	136	170	3	6.3		
		"	" " " " 1100F-2HR "	132	168	1	8.6		
		"	" " " " 4HR "	132	168	5	11.5		

TABLE J21 (Continued)

MECHANICAL PROPERTIES OF HEAT TREATED EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Heat Treatment	Yield Strength KSI (0.2% Offset)	Ultimate Tensile Strength KSI	Elongation in 4D Per Cent	Reduction in Area Per Cent	Notch Tensile KSI	Brinell Hardness Number
29		P359	As Cast	111	150	7.5	14.0	219	
		"	1500F-2HR-WQ+1000F-8HR-AC	149	165	5	11.5	217	
		"	" " " 1100F-4HR "	144	158	3	6.7	226	
		"	" " " 1050F-6HR "	150	163	4	8.1	222	
		"	1450 " " " 1000F-8HR "	148	163	4	10.6	214	
		"	" " " 1100F-4HR "	144	153	3	7.0	224	
		"	" " " 1050F-6HR "	146	160	5	6.7	221	
		P149	1550F-2HR-WQ+1000F-4HR-AC	158	163.6	2	4.9		363
		"	" " " " -8HR "	159	168.0	2	4.0		352
		"	" " " 1100F-2HR "	155	170.0	3	7.0		352
		"	" " " 4HR "	152	162.4	4	7.4		352
		"	1500F " " " 1000F-4HR "	134	165.0	2	4.9		352
		"	" " " " 8HR "	136	170.0	3	6.3		352
		"	" " " " 1100F-2HR "	132	166.0	3	8.6		341
		"	" " " " 4HR "	132	166.0	5	11.5		352
32	15V	P146-2A	As Cast	115.0	128.8	3	9.2	(P146-3) 122.8	293
		B	As Cast	115.0	134.0	5	11.5	167.2	
		C	As Cast	115.0	135.2	6	18.6	177.6	
		P146	1350F-2HR-WQ+900F-2HR-AC	141.8	144.0	4	1.8		
		"	" " " " 4HR "	164.0	166.5	3	3.8		
		P146	" " " " 8HR "	135.3	136.7	2	3.3		
41	2Cu	P523	As Cast	85	106	15	25.9	156	
		"	" " " 1000F-8HR-AC						
		"	" " " " 24HR "						
		"	" " " 900F-72HR "	88	105	15	23.9		241
		"	" " " " 24 "	88	104	16	22.4		235
		"	" " " 48 "	89	106	15	24.6		248
		"	" " " 800F-8HR "	84	104	12	20.4		299
		"	" " " " 24HR "	88	104	15	24.4		241
		"	" " " 48HR "	88	105	16	27.2		241
		"	" " " 700F-8HR "	86	104	16	25.9		235
		"	" " " " 24HR "	86	104	17	26.5		223
		"	" " " 48HR "	88	105	16	30.5		223
		"	" " " 600F-24HR "						
		"	" " " " 48HR "						
		P202	1475F-2HR-WQ+1200F-2HR-AC	55	82.8	21	41.4		170
		"	" " " " 8HR "	58	80	19	38.1		183
		P579	As Cast	100	117	14	18.9	173	255
		"	" " " 1000F-80HR-AC	100	118	13	19.7	174	255
		"	" " " 900F-8HR "	102	110	13	18.1		262
		"	" " " 600F-8HR "	98	115	6	17.5		262
		"	" " " " 80HR "	102	118	13	9.2		269
		"	" " " " 48HR "	102	118	13	22.4		248
42	6Al-2Cu	P223-1A	As Cast	111	123.6	8	15.8	182.4	283
		B	As Cast	110	122.2	9	14.9	182.8	
		C	As Cast	111	122.4	9	15.0	182.8	
		P223	1475F-2HR-WQ+1200F-2HR-AC	108	118.8	10	23.2		262
		"	" " " " 8HR "	106	116.8	10	23.2	(P182-3) 182.4	269
44	3Al-2V	P144-1A	As Cast	94.0	106.8	11	28.5	168.4	253
		B	As Cast	94.0	108.8	11	29.2		
		C	As Cast	94.0	108.0	12	29.9		
		P144	As Cast	94.0	107.6	11	29.4		
		"	1450F-2HR-WQ+900F-2HR-AC	101.6	113.8	9	13.2		
		"	" " " " 4HR "	105.3	113.9	5	15.3		
		"	" " " " 8HR "	105.4	112.5	9	21.2		
		"	1350F " " " 2HR "	101.4	108.6	12	30.3		
		"	" " " " 4HR "	102.0	106.6	10	21.1		
		"	" " " " 8HR "	100.2	106.5	14	30.4		
		"	" " " " 16HR "	102.2	108.9	9	21.2		
45	8Al-8Fe	P224-1A	As Cast	112	129.4	12	30.4	(P112-2) 129.4	286
	10b-11a	B	As Cast	113	125.7	14	27.7	152	
		C	As Cast	114	126.4	14	25.4	196.8	
		P224	1000F-4HR-AC	114	127.7	14	12.4		277
		"	" " " " 8HR "	120	127.8	5	14.5		265
		"	" " " " 16HR "	110	125.4	6	13.4		251

TABLE J21 (Continued)

MECHANICAL PROPERTIES OF HEAT TREATED EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Heat Treatment	Yield Strength KSI (0.2% Offset)	Ultimate Tensile Strength KSI	Elongation In 4D Per Cent	Reduction In Area Per Cent	Notch Tensile KSI	Brinell Hardness Number
46	3Al-5Be	P182-2A	As Cast	68	92.0	14	25.2	127.2	201
"	"	"	B As Cast	68	90.4	13	26	132	"
"	"	"	C As Cast	68	91.6	13	23	133.2	"
"	"	P182	1500F-2HR-MQ	78	126.0	4.5	9.2	"	293
"	"	"	" " " + 900F-1HR-AC	78	106.0	3	8.6	"	277
"	"	"	" " " " 2HR "	72	108	4.5	10	"	248
"	"	"	" " " " 4HR "	72	115.6	7	11.9	"	248
"	"	"	" " " " 8HR "	60	113.2	7	11.9	"	241
"	"	"	" " " " 800F-1HR "	81	172	2.5	4	"	285
"	"	"	" " " " 2HR "	78	124.8	3	6.3	"	293
"	"	"	" " " " 4HR "	76	123.6	6	7.6	"	293
"	"	"	" " " " 8HR "	76	123.2	4	7.4	"	285
49	6Al-6V-2Sn	P234-1A	As Cast	111.0	132.0	10	32.5	(P234-2) 206.0	296
"	"	"	B As Cast	115.0	136.0	8	21.0	204.0	"
"	"	"	C As Cast	111.0	131.0	9	29.0	204.4	"
"	"	P234	1600F-2HR-MQ+1000F-2HR-AC	---	163.6	2	3.8	"	363
"	"	"	" " " " 4HR "	150	163.6	2	3.8	"	352
"	"	"	" " " " 1100F-2HR "	146	158.8	2	7.0	"	352
"	"	"	" " " " 4HR "	150	160.0	1	1.6	"	352
"	"	"	" " " " 1000F-2HR "	142	159.2	2	4.9	"	352
"	"	"	" " " " 4HR "	145	157.2	1	0.3	"	352
"	"	"	" " " " 1100F-2HR "	141	155.2	2	7.0	"	341
"	"	"	" " " " 4HR "	140	154.2	3	7.6	"	341
50	13Sn-2Al-10Zr	P230-1A	As Cast	122.0	142.0	4	5.3	196.2	305
"	"	"	B As Cast	123.0	143.2	11	18.9	186.0	"
"	"	"	C As Cast	123.0	144.0	10	18.1	190.0	"
"	"	P230	1075F-4HR-AC	126	141.6	7	11.9	"	293
"	"	"	" 16HR "	124	143.6	4	7.6	"	311
55	6Al-4V-0.5W	P297-1A	As Cast	104.0	124.0	9	25.7	"	277
"	"	"	B As Cast	106.0	126.0	10	25.2	"	"
"	"	"	C As Cast	104.0	124.0	12	41.4	"	"
"	"	313	"	"	"	"	"	200 197.6 203.2	243
55	"	313	1600F-2HR-MQ+1000F-2HR-AC	121	139.6	5	11.5	"	321
"	"	"	" " " " 4HR "	120	139.6	8	15.1	"	311
"	"	"	" " " " 1100F-2HR "	123	136.7	6	13.5	"	286
"	"	"	" " " " 4HR "	119	134.0	8	16.7	"	293
"	"	"	" " " " 1000F-2HR "	121	138.0	7	13.5	"	302
"	"	"	" " " " 4HR "	120	138.8	7	12.9	"	302
"	"	"	" " " " 1100F-2HR "	116	132.8	8	12.7	"	293
"	"	"	" " " " 4HR "	116	133.6	9	17.5	"	311
56	6Al-4V-0.5Ta	348	As Cast	104	124	11	38.9	192	282.3
"	"	"	As Cast	108	128	11	24.9	190	"
"	"	"	As Cast	106	124	11	33.6	190	"
"	"	"	1600F-2HR-MQ+ 900F-2HR-AC	117	140.4	6	10.0	222	322
"	"	"	" " " " 8HR-AC	123	140.0	4	12.2	222	321
"	"	"	" " " " 1000F-2HR "	114	138.2	9	25.2	"	311
"	"	"	" " " " 4HR "	119	136.8	9	25.0	"	321
"	"	"	" " " " 1100F-2HR "	117	134.0	3	14.5	"	302
"	"	"	" " " " 4HR "	120	133.0	5	15.9	"	302
"	"	"	" " " " 1HR " 1000F-2HR "	120	136.0	4	12.2	"	311
"	"	"	" " " " 4HR "	120	136.8	6	12.9	"	311
"	"	"	" " " " 1100F-2HR "	118	134.2	6	7.6	"	302
"	"	"	" " " " 4HR "	116	131.2	7	14.2	"	311

TABLE J-1 (Continued)

MECHANICAL PROPERTIES OF HEAT TREATED EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Heat Treatment	Yield Strength KSI (0.2% Offset)	Ultimate Tensile Strength KSI	Elongation In 4D Per Cent	Reduction In Area Per Cent	Notch Tensile KSI (148-3)	Brinell Hardness Number
57	3Al-7Mn -0.25Si	P298-1A	As Cast	114	134	6	10.6	192	285
"	"	"	B As Cast	113	134	8	12.2	189.6	"
"	"	"	C As Cast	112	133.2	7	10.6	192	"
"	"	P29d	1450F-2HR-WQ	114	132	10	18.9	"	269
"	"	"	" " " 1000F-1HR-AL	Not	148	0	0	"	388
"	"	"	" " " 2HR "	Taken	156	0	0	"	375
"	"	"	" " " 4HR "	"	152	0	0	"	388
"	"	"	" " " 8HR "	"	156	0	0	"	375
"	"	"	" " " 800F-1HR "	"	156	0	0	"	388
"	"	"	" " " 2HR "	"	152.8	0	0	"	388
"	"	"	" " " 4 HR "	"	158.4	0	0	"	388
"	"	"	" " " 2HR "	"	162.4	0	0	"	388
60	4Al-4Sn-8Zr -1Fe-1Cr-1V	P300	As Cast	133	152	10	21	"	"
"	"	"	1025F-4HR-AC	135	152	11	19.4	"	341
"	"	"	" 8HR "	130	146	3	4.9	"	311
"	"	"	" 16HR "	137	153	8	10.6	"	331
"	"	P489	1000F-8HR "	157	174	3	5.2	"	"
"	"	"	" 2HR "	160	176	4	4.3	"	"
"	"	"	" 4HR "	161	176	3	2.4	172	"
"	"	"	As Cast	151	170	7	10.5	"	"
"	"	"	500F-8HR "	"	"	"	"	"	"
"	"	"	" 2HR "	156	177	4	7	"	352
"	"	"	" 4HR "	156	156	0	0	161	352
"	"	"	" 2HR "	156	177	4	5.3	"	352
"	"	"	800F-8HR "	155	173	7	11.5	"	"
"	"	"	" 2HR "	157	173	2	2.4	"	"
"	"	"	" 4HR "	160	182	5	7.6	"	"
60	4Al-4Sn P300-1A 87r-1Fe 1V-1Cr	A As Cast	132.0	151.7	9	21.5	223.2	321	"
"	"	B As Cast	132.0	152.0	10	22.4	218.8	"	"
"	"	C As Cast	136.0	154.0	10	19.7	223.2	"	"
"	"	P489	As Cast	151.0	170.0	7	10.5	"	"
"	"	300	1450F-2HR-WQ-1000F-1HR-AC	152	178.8	3	10.9	"	363
"	"	"	" " " 2HR "	153	176.8	4	10.0	"	363
"	"	"	" " " 4HR "	151	173.6	3	5.3	"	363
"	"	P300	As Cast	133	152	10	21.0	221	"
"	"	P489	As Cast	151	170	7	10.5	231	"
"	"	P300	1450F-2HR-WQ-1000F-1HR-AC	---	171	2	2.4	"	"
"	"	"	" " " 2HR "	156	160	2	3.3	"	"
"	"	"	1500F " " " 4HR "	154	179	2	5.3	"	"
"	"	"	1450F " " " 1HR "	152	179	3	10.9	"	"
60	"	P300	1450F-2HR-WQ-1000F-2HR-AC	153	177	4	10.9	"	"
"	"	"	" " " 4HR "	151	174	3	5.3	"	"
"	"	"	1500F " " " 8HR "	154	176	3	3.3	"	"
"	"	P489	" " " " " "	173	177	1	0.7	16d	"
"	"	P300	1500F-2HR-WQ-1000F-1HR-AL	---	180.8	2	2.4	"	375
"	"	"	" " " 2HR "	156	160.0	2	3.3	"	363
"	"	"	" " " 4HR "	154	178.8	2	5.3	"	375
63	7Al-2Mn 0.5Mn-0.51a	P350	As Cast	155	172	1	3.3	128	361
"	"	"	As Cast	155	162.4	1	1.6	128	"
"	"	"	As Cast	150	170.3	1	4	152	"
"	"	"	1000F 2HR-AC	Not	169.6	2	4.2	"	363
"	"	"	" 4HR "	Taken	178	2	4.2	"	363
"	"	"	1200F-2HR-AC	"	146	1	1.6	"	352
"	"	"	" 4HR "	"	162.4	1	2.4	"	365
64	4Al-4Sn-8Zr -1Fe-1Cr-1V	P300 A	As Cast	124	141.2	11	22.4	213.2	290
"	"	" B	As Cast	126	142	10	21.5	214	"
"	"	" C	As Cast	125	141.8	10	23	214.4	"
"	"	P300	1450F-2HR-WQ-1000F-1HR-AL	128	140.4	10	16.9	"	293
"	"	"	" " " 2HR "	128	141.2	7	10.5	"	293
"	"	"	" " " 4HR "	124	138.8	8	10.8	"	293

TABLE J21 (Continued)

MECHANICAL PROPERTIES OF HEAT TREATED EXPERIMENTAL ALLOYS

Alloy No.	Alloy Type	Heat No.	Heat Treatment	Yield Strength	Ultimate Tensile Strength	Elongation		Reduct. of Area		Notch Tensile	Brinell Hardness Number
				KSI (0.2% Offset)	KSI	In 4D Per Cent	In Area Per Cent	Per Cent	Per Cent	KSI	
69	6Al-4V 1Cr-1Mo	434	As Cast	129	147.6	9	15.1			204	317
		"	As Cast	126	144.8	7	18.6				
		"	As Cast	128	145.0	8	17.0			204	
		None-A B	As Cast	130	152.0	7	15.1				321
			As Cast	130	152.0	6	9.2				
		434	1550F-2HR-WQ+1000F-4HR-AC	144	159.2	2	6.7				341
		"	1550F-2HR-WQ+1000F-8HR-AC	144	158.8	2	6.7				341
		"	1550F-2HR-QW+1100F-2HR-AC	140	154.0	4	10.5				331
		"	1550F-2HR-QW+1100F-4HR-AC	141	154.4	4	7.0				352
		"	1500F-2HR-QW+1000F-4HR-AC	145	160.0	4	7.6				341
		"	1500F-2HR-WQ+1000F-8HR-AC	141	156.8	3	7.6				331
		"	1500F-2HR-WQ+1100F-2HR-AC	139	154.4	4	9.7				321
		"	1500F-2HR-WQ+1100F-4HR-AC	136	150.0	5	11.2				311
105	8Zr-4Al- 4Sn-1.5Fe- 1.5Cr-1.5V	596	As Cast	154	171	4	5.3			216	352
		"	1500F-2HR-AC	159	175	3	6.5				363
		"	1400F-2HR-AC	154	168	8	12.9				352
		"	1300F-2HR-AC	154	168	4	6.7				352
		"	1200F-2HR-AC	158	164	2	1.6				352
		"	1400F-2HR-AC+1000F-48HR-AC		142	0	0				
		"	1400F-2HR-AC+ 900F-48HR-AC		184	0.7					
		"	1400F-2HR-AC+ 800F-48HR-AC		136	1	1.6				401
		"	1400F-2HR-AC+ 700F-48HR-AC		101	0	0				401
		"	1400F-2HR-AC+ 600F-48HR-AC		170	4	8.6				352
		"	As Cast	132	148	10	20.0				321
106	8Zr-4Al- 4Sn-.5Fe- .5Cr-.5V	597	As Cast	132	148	10	20.0				321
		"	1500F-2HR-WQ+1000F-2HR-AC	146	162	4	9.2				352
		"	1400F-2HR-WQ+1000F-2HR-AC	144	158	5	6.3				331
		"	1000F-2HR-AC	135	149	4	14.0				311
		"	1000F-48HR-AC	138	151	5	8.6				321
		"	900F-48HR-AC	138	152	8	18.9				341
		"	800F-48HR-AC	138	154	7	13.5				331
		"	700F-48HR-AC	136	152	8	17.5				311
		"	600F-48HR-AC	133	140	10	18.9				293
		"	As Cast	132	148	10	22.4			219	321
		"	As Cast	132	149	11	16.7				321
		"	1500F-2HR-WQ+1000-2HR-AC	146	162	4	9.2				352
		"	1400F-2HR-WQ+1000-2HR-AC	144	158	5	6.3				331
		"	As Cast		168	1	1.6			140	
107	10Cr	606	As Cast		172	1	3.3				
		"	1400F-2HR-WQ		150	1	0.7				331
		"	1400F-2HR-WQ+800F-1HR-AC		107	0	0				429
		"	1400F-2HR-WQ+800F-2HR-AC		176	0	0				429
		"	1400F-2HR-WQ+800F-4HR-AC		173	0	0				
		"	As Cast		152	11	20.0			208	331
108	8Zr-4Al- 4Sn-2V	607	As Cast	130	152	11	20.0				331
		"	900F-48HR-AC	140	157	9	16.7				331
		"	800F-48HR-AC	139	158	10	18.9				331
		"	700F-48HR-AC	138	159	9	19.1				331
		"	600F-48HR-AC	135	158	9	23.2				331
		"	1000F-48HR-AC	140	142	1	5.3				331
		"	1500F-2HR-WQ+1000-1HR-AC	145	161	7.5	12.9				352
		"	1500F-2HR-WQ+1000-2HR-AC	145	161	7	12.9				341
		"	1500F-2HR-WQ+1000-4HR-AC	144	160	8	15.1				352
		"	1500F-2HR-WQ+1000-8HR-AC	147	164	8	17.0				352
		"	1500F-2HR-WQ+ 900-1HR-AC	144	162	6	11.2				352
		"	1500F-2HR-WQ+ 900-2HR-AC	146	162	6	12.9				352
		"	1500F-2HR-WQ+ 900-4HR-AC	146	163	9	15.1				352
		"	1500F-2HR-WQ+ 900-8HR-AC	144	161	7	14.2				352
		"	As Cast		152	11	20.0			208	331

TABLE J22

HEAT ANALYSES

Heat No.	Per Cent																		BHA
	C	H	O	N	Fe	Al	V	Co	Cr	Cu	Pg	Mn	Si	Pb	Mo	Sn			
P1	.004	.0033	.16	.016	.261	5.02	3.92	.002	.02	.005	.005	.004	.03	.002	.01	.015	.02	321	
P2	.008	.003	.133	.017	.275	6.01	3.92	.002	.015	.005	.005	.03	.03	.002	.015	.015	.03	326	
P3	.057	.0022	.175	.016	.268	5.98	3.99	.002	.03	.005	.005	.004	.04	.002	.10	.008	.004	321	
P4	.025	.0021	.124	.017	.212	5.91	3.96	.002	.04	.005	.005	.01	.08	.002	.10	.10	.015	311	
P5	.022	.003	.1466	.016	.167	5.98	4.13	.002	.10	.005	.005	.004	.08	.002	.10	.04	.01	305	
P6	.031	.004	.1515	.016	.210	6.12	4.08	.002	.015	.005	.005	.008	.04	.002	.02	.015	.008	315	
P7	.045	.003	.1425	.014	.210	6.02	4.04	.002	.02	.02	.005	.004	.03	.002	.02	.02	.015	315	
P8	No Analysis																		
P9	.031	.004	.1755	.021	.220	6.07	4.04	.002	.015	.005	.005	.008	.015	.002	.01	.015	.01	315	
P10	.070	.0034	.2275	.025	.268	5.93	3.94	.002	.015	.005	.005	.004	.02	.002	.015	.02	.015	326	
P11	.056	.0031	.215	.025	.270	5.82	3.98	.002	.01	.005	.005	.008	.015	.002	.01	.008	.008	315	
P12	.030	.0031	.213	.024	.255	6.13	4.30	.002	.02	.005	.005	.005	.03	.002	.02	.015	.01	361	
P13	.036	.0026	.15	.018	.205	5.98	4.15	.002	.02	.005	.005	.004	.03	.002	.02	.015	.004	304	
P14	.052	.0026	.19	.023	.215	5.85	4.15	.002	.02	.005	.005	.004	.03	.002	.02	.015	.006	361	
P15	.027	.0037	.1675	.018	.180	6.10	3.98	.002	.02	.01	.005	.004	.04	.002	.04	.02	.02	325	
P16	.051	.0028	.2180	.022	.215	5.86	4.11	.002	.02	.005	.005	.004	.04	.002	.03	.02	.03	324	
P17	.028	.0025	.305	.016	.180	6.16	4.05	.002	.02	.005	.005	.01	.04	.002	.02	.02	.003	302	
P18	.042	.0032	.165	.021	.145	6.07	4.05	.002	.02	.005	.005	.008	.04	.002	.02	.03	.01	321	
P19	.070	.0034	.1858	.020	.215	6.03	2.97	.002	.02	.005	.005	.002	.07	.002	.10	.02	.006	296	
P20	.355	.0025	.335	.022	.195	6.73	3.88	.002	.02	.006	.005	.002	.02	.002	.10	.015	.004	314	
P21	.050	.0036	.186	.019	.180	5.81	3.94	.002	.10	.005	.005	.015	.08	.002	.10	.015	.01		
P22	See p24 - P22 Mismatch																		
P23	.042	.0043	.2698	.022	.210	5.55	3.75	.002	.02	.015	.005	.002	.03	.002	.10	.03	.004	308	
P24	.010	.0024	.162	.020	.135	3.65	3.72	.002	.01	.005	.005	4.18	.005	.002	.02	.002	.01	311	
P25	.077	.004	.22	.023	.320	5.66	3.94	.002	.02	.025	.005	.01	.03	.002	.10	.02	.004		
P26	.067	.0024	.23	.014	.170	4.53	3.50	.002	.01	.005	.005	.008	.005	.002	.01	.02	2.35	308	
P27	.054	.0027	.2575	.014	.235	5.70	3.66	.002	.02	.008	.005	.004	.03	.002	.03	.002	.01	318	
P28	.060	.0023	.245	.022	.225	5.67	3.73	.002	.015	.005	.005	.01	.03	.002	.015	.015	.008		
P29	.066	.0030	.2275	.023	.236	5.92	3.76	.0005	.015	.005	.005	.008	.03	.002	.02	.015	.01	321	
P30	.073	.0026	.2925	.027	.240	6.04	3.92	.0005	.02	.008	.005	.008	.03	.002	.02	.015	.008	334	
P31	.051	.0021	.18	.023	.300	6.11	3.92	.0005	.02	.005	.005	.008	.02	.002	.02	.015	.015	318	
P32	.075	.003	.355	.025	.240	5.93	3.86	.0005	.02	.01	.005	.008	.03	.002	.015	.015	.01	341	
P33	.070	.0032	.256	.015	.175	6.71	3.015	.002	.03	.005	.005	.004	.005	.002	.10	.005	2.34	314	
P34	.040	.0026	.1975	.02	.060	5.77	3.88	.002	.02	.01	.005	.006	.02	.001	.02	.015	.015	315	
P35	.053	.0022	.185	.027	.260	6.03	3.97	.002	.015	.005	.005	.015	.015	.002	.015	.015	.03	321	
P36	.062	.0041	.2125	.027	.245	6.02	3.87	.002	.02	.005	.005	.004	.03	.002	.10	.02	.01	331	
P37	.079	.0033	.2625	.026	.320	5.73	3.76	.002	.02	.005	.005	.004	.02	.002	.10	.015	.006	334	
P38	(13V-11Cr-2.5Al)																		
P39	.020	.0046	.1438	.017	.226	2.62	12.9	.002	11.3	.005	.005	.002	.04	.002	.04	.03	.002	277	
P40	.098	.0025	.2463	.029	.370	6.01	3.82	.002	.02	.005	.005	.004	.02	.002	.10	.015	.008		
P41	.092	.0042	.2512	.027	.420	5.96	3.82	.002	.10	.005	.005	.006	.02	.002	.08	.015	.01	334	
P42	.096	.0032	.226	.026	.340	5.95	3.73	.002	.02	.006	.005	.004	.015	.002	.10	.02	.01	316	
P43	.064	.0034	.16	.021	.310	6.04	3.79	.002	.02	.006	.005	.01	.01	.002	.10	.015	.015	299	
P44	.060	.0030	.229	.026	.260	5.87	3.64	.002	.02	.005	.005	.004	.01	.002	.10	.005	.006	331	
P45	.030	.0025	.143	.020	.175	6.71	12.7	.002	9.006	.005	.005	.002	.015	.002	.02	.025	.004	262	
P46	(13V-11Cr-2.5Al)																		
P47	.076	.0039	.162	.022	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P48	(13V-11Cr-2.5Al)																		
P49	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P50	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P51	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P52	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P53	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P54	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P55	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P56	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P57	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P58	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P59	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	
P60	.067	.0032	.180	.02	.38	6.04	3.79	.002	.02	.006	.005	.004	.02	.002	.1	.012	.013	334	

TABLE J22 (Continued)

HEAT ANALYSES

Heat No.	C	II	O	N	Fe	Al	V	Cu	Cr	Mn	Mg	Mo	Ni	Pb	Si	Co	Sn	BHN
(13V-11Cr-4Al)																		
P52	.024	.0063	.153	.018	.21	4.05	13.5	.015	10.1	.005	.005	.003	.015	.002	.02	.02	.002	291
P53	.075	.0025	.185	.028	.34	6.01	3.93	.007	1	.005	.005	.004	.32	.002	.1	.01	.01	324
P54	.080	.0029		.031	.32	6.07	3.92											324
P55	.028	.0049	.153	.022	.19	1.58	3.0	.007	10.0	.005	.005	.007	.02	.002	.015	.01	.002	283
(13V-11Cr-13Al)																		
P56	.048	.0026		.028	.22	6.23	3.98											321
P57	.054	.0024	.20	.024	.29	6.05	4.01	.005	1	.005	.005	.01	.015	.002	.1	.001	.012	321
P58	.081	.0032	.193	.012	.40	6.05	3.97	.002	1	.005	.005	.01	.01	.002	.02	.01	.01	
P59	.027	.0017	.185	.017	.20	.01	.005	.002	.015	.005	.005	.002	.005	.002	.01	.002	.003	
P60	.029	.0025	.175	.014	.21	.01	.005	.002	.02	.005	.005	.002	.005	.002	.01	.002	.003	
(C.P. 11)																		
P61	.032	.0044	.158	.027	.20	1.65	13.5	.002	10.7	.005	.005	.002	.025	.002	.03	.015	.002	280
(13V-11Cr-14Al)																		
P62	.028	.0045	.146	.027	.20		12.9	.002	10.3	.005	.005	.002	.04	.002	.02	.002	.002	260
(13V-11Cr)																		
Ti6Al-4V																		
P63	.070	.0025	.18	.030	.29	5.87	4.83	.007	1	.02	.005	.003	.01	.002	.08	.008	.015	321
Ti7Al-3Mo																		
P64	.020	.0032	.131	.0083	.12	7.37	.005	.002	.025	.005	.005	.002	.005	.002	.02	3.33	.002	308
Ti53Al-53V-.45Fe-25Sn-.25Cu																		
P65	.019	.0023	.10	.010	.59	5.62	5.46	.002	.02	.22	.005	.002	.01	.002	.03	.015	1.97	311
Ti33V-11Cr																		
P66	.023	.0023	.19	.028	.20	.01	12.9	.002	9.5	.005	.005	.002	.03	.002	.01	.002	.003	288
Ti6Al-4V																		
P67	.038	.0036	.16	.022	.32	6.08	3.93	.002	1	.02	.005	.015	.02	.002	.015	.015	.02	
Ti7Al-3Mo																		
P68	.024	.0033	.10	.010	.13	7.45	1	.002	.025	.005	.005	.002	.005	.002	.03	3.07	.002	315
Ti53Al-53V-.45Fe-25Sn-.25Cu																		
P69	.023	.0028	.12	.011	.59	5.59	5.41	.002	.02	.22	.005	.003	.01	.002	.02	.015	1.99	324
Ti6Al-4V																		
P70	.060	.0034	.14	.033	.30	5.99	4.32	.002	1	.005	.005	.015	.02	.002	.015	.015	.02	337
P71	.082	.0033	.19	.033	.32	5.98	4.13	.002	1	.005	.005	.01	.015	.002	.02	.015	.02	337
P72	.084	.0033	.13	.032	.28	6.14	4.07	.002	1	.005	.005	.01	.015	.002	.02	.015	.03	
P73	.094	.0033	.19	.036	.31	5.97	4.27	.002	1	.005	.005	.05	.015	.002	.02	.015	.03	392
P74	.072	.0026	.17	.031	.28	6.26	4.19	.002	1	.005	.005	.004	.015	.002	.05	.02	.02	
P75	.054	.0029	.17	.027	.19	6.14	3.95	.002	1	.005	.005	.008	.02	.002	.025	.01	.01	
P76	.072	.0033	.19	.032	.32	6.20	4.17	.002	1	.005	.005	.01	.015	.002	.025	.02	.04	
P77	.079	.0029	.17	.030	.24	6.16	4.15	.002	1	.01	.005	.008	.015	.002	.05	.02	.03	
P78	.072	.0020	.16	.032	.21	6.17	4.45	.002	1	.01	.005	.004	.015	.002	.02	.02	.04	
P79	.076	.0032	.20	.032	.24	6.34	4.17	.002	1	.01	.005	.004	.02	.002	.1	.02	.03	
P80	.071	.0033	.23	.032	.25	6.35	4.12	.002	.06	.005	.005	.004	.015	.002	.015	.015	.02	
P81	.105	.0026	.21	.034	.24	6.17	4.17	.002	1	.01	.005	.004	.015	.002	.02	.02	.03	
P82	.090	.0029	.19	.032	.18	6.31	4.15	.002	1	.01	.005	.008	.015	.002	.02	.03	.02	
P83	.100	.0024	.23	.032	.27	6.24	4.15	.002	1	.01	.005	.005	.015	.002	.02	.02	.02	
P84	.094	.0026	.24	.033	.26	6.04	3.97	.002	1	.005	.005	.01	.01	.002	.015	.02	.02	
P85	.094	.0030	.25	.037	.28	6.09	3.90	.002	1	.005	.005	.01	.02	.002	1	.015	.03	
P86	.092	.0026	.22	.037	.24	5.65	3.66	.002	1	.005	.005	.01	.01	.002	.02	.02	.02	
P87	.109	.0034	.22	.033	.23	6.05	3.76	.002	.04	.01	.005	.01	.015	.002	.02	.02	.03	
P88	.091	.0026	.22	.034	.27	6.24	4.38	.002	1	.005	.005	.005	.015	.002	.02	.02	.02	
P89	.078	.0033	.21	.030	.24	6.07	3.72	.002	.02	.005	.005	.01	.01	.002	.02	.02	.02	
P90	.104	.0038	.22	.035	.24	6.00	3.64	.002	.01	.01	.005	.004	.01	.002	.015	.015	.03	

TABLE J22 (Continued)

HEAT ANALYSES

Heat No.	C	H	O	N	Fe	Al	V	Co	Cr	Cu	Mg	Mn	Ni	Pb	Si	Mo	Sn	BHN
P91	No Analysis																	
P92	.094	.0041	.24	.037	.27	6.19	3.96	.002	.1	.005	.005	.002	.005	.002	.015	.004	.02	
P93	.002	.0039	.26	.037	.24	6.18	3.82	.002	.1	.005	.005	.002	.005	.002	.015	.005	.02	
P94	.041	.0031	.15	.021	.28	5.94	4.06	.002	.1	.01	.005	.003	.03	.002	.025	.015	.01	
P95	.038	.0033	.14	.017	.24	6.74	3.83	.002	.04	.015	.005	.003	.04	.002	.02	.02	.01	
P96	.066	.0026	.16	.019	.26	6.24	3.99	.002	.08	.005	.005	.003	.03	.002	.015	.015	.015	334
P97	.058	.0030	.14	.018	.25	6.06	3.92	.002	.03	.01	.005	.003	.02	.002	.015	.015	.01	
P98	.094	.0035	.26	.039	.27	6.07	4.03	.002	.1	.005	.005	.003	.015	.002	.015	.01	.015	355
P99	.088	.0017	.28	.037	.25	5.95	3.80	.002	.1	.04	.005	.004	.015	.002	.02	.015	.02	
P100	.093	.0009	.26	.034	.23	5.94	4.03	.002	.1	.01	.005	.004	.02	.002	.025	.015	.03	348
P101	.041	.0036	.14	.019	.25	5.96	3.97	.002	.06	.01	.005	.003	.03	.002	.025	.015	.015	
P102	.094	.0036	.20	.022	.25	5.96	3.87	.002	.1	.01	.005	.003	.03	.002	.03	.015	.015	345
P103	.035	.0031	.24	.015	.25	5.95	4.06	.002	.05	.02	.005	.003	.04	.002	.03	.02	.01	
P104	.072	.0031	.20	.028	.26	6.00	3.97	.002	.1	.01	.005	.01	.03	.002	.02	.015	.02	348
P105	.086	.0031	.23	.032	.20	5.90	3.93	.002	.1	.01	.005	.004	.02	.002	.015	.02	.03	
P106	.049	.0028	.16	.022	.25	5.98	4.02	.002	.1	.01	.005	.003	.03	.002	.05	.02	.01	311
P107	.092	.0039	.21	.031	.26	5.88	3.97	.002	.1	.01	.005	.01	.03	.002	.02	.02	.02	337
P108	.094	.0030	.21	.030	.27	5.89	3.85	.002	.1	.01	.005	.003	.015	.002	.02	.015	.015	337
P109	.108	.0033	.19	.032	.27	5.97	3.98	.002	.1	.01	.005	.004	.03	.002	.002	.02	.03	
P110	.048	.0031	.14	.016	.24	6.03	3.82	.002	.03	.08	.005	.004	.004	.002	.02	.03	.01	
P111	.094	.0046	.22	.028	.27	5.96	3.96	.002	.05	.03	.005	.003	.02	.002	.015	.015	.015	
P112	.070	.0034	.22	.026	.26	6.05	3.90	.002	.1	.01	.005	.003	.03	.002	.02	.02	.015	
P113	.080	.0037	.18	.030	.26	5.99	3.95	.002	.1	.01	.005	.003	.03	.002	.02	.02	.02	
P114	.080	.0028	.15	.026	.27	5.85	3.83	.002	.1	.01	.005	.003	.02	.002	.03	.02	.015	
P115	.042	.0026	.19	.025	.23	5.95	4.02	.002	.015	.015	.005	.002	.01	.002	.02	.015	.02	317
P116	.060	.0031	.21	.028	.20	5.92	3.95	.002	.015	.01	.005	.003	.02	.002	.025	.03	.03	
P117	.096	.0029	.23	.033	.26	5.88	3.76	.002	.1	.01	.005	.003	.02	.002	.02	.005	.015	
P118	.098	.0038	.26	.032	.28	5.99	3.99	.002	.1	.005	.005	.003	.04	.002	.1	.015	.005	
P119	.042	.0022	.11	.028	.27	5.90	3.78	.002	.08	.01	.005	.002	.03	.002	.02	.02	.008	334
P120	.058	.0026	.15	.022	.18	5.94	4.03	.002	.1	.008	.005	.002	.04	.002	.1	.015	.015	
P121	.100	.0036	.28	.035	.29	5.96	3.96	.002	.1	.002	.005	.002	.04	.002	.1	.015	.015	
P123	.068	.0031	.20	.029	.26	6.02	3.99	.002	.05	.005	.005	.002	.04	.002	.1	.015	.005	
P124	.048	.0029	.18	.025	.19	6.05	4.04	.002	.02	.008	.005	.002	.03	.002	.1	.015	.015	
P125	.076	.0036	.21	.032	.28	5.97	3.98	.002	.1	.01	.005	.003	.04	.002	.1	.03	.015	
P126	.078	.0036	.20	.038	.25	5.83	3.84	.002	.03	.015	.005	.002	.03	.002	.03	.02	.015	
P127	.058	.0032	.22	.030	.21	5.94	3.89	.002	.1	.04	.005	.003	.04	.002	.1	.03	.015	
P128	.074	.0038	.28	.031	.25	5.94	3.88	.002	.03	.1	.005	.002	.03	.002	.03	.02	.015	
P129	.065	.0027	.15	.025	.22	5.93	3.92	.002	.03	.04	.005	.003	.03	.005	.03	.02	.008	334
P131	.026	.0018	.15	.024	.21	6.03	3.90	.002	.015	.01	.005	.003	.015	.002	.02	.015	.002	317
P132	.059	.0028	.212	.032	.21	6.05	4.05	.002	.04	.01	.005	.003	.03	.002	.025	.02	.01	331
P133	.044	.0029	.23	.036	.19	5.82	4.09	.002	.025	.01	.005	.002	.02	.002	.03	.02	.01	
P134	.066	.0034	.226	.034	.23	6.05	4.03	.002	.02	.005	.005	.002	.01	.002	.02	.03	.01	
P135	.053	.0034	.31	.029	.18	6.00	3.84	.002	.07	.008	.005	.003	.015	.002	.02	.02	.008	321
P136	No Four - Electrode Broken																	
P137	.036	.0015	.174	.016	.12	6.10	3.94	.002	.015	.005	.005	.003	.01	.002	.02	.02	.002	311
P138	No Four - Electrode Broken																	
(T18V-5Fe)																		
P139	.022	.0031	.101		5.59	.01	8.03	.002	.015	.01	.005	.002	.03	.002	.015	.002	.003	
(T18V-5Fe-1Al)																		
P140	.030	.0024	.096	.014	5.4	.79	8.15	.002	.02	.005	.005	.002	.03	.002	.015	.005	.002	
P141	.042	.0024	.152	.028	.29	5.85	4.05	.002	.03	.01	.005	.003	.03	.002	.08	.006	.004	
P142	.100	.0036	.196	.035	.23	6.05	4.09	.002	.04	.008	.005	.002	.02	.002	.02	.02	.01	341
P143	.076	.003	.25	.029	.23	5.90	3.90	.002	.01	.1	.005	.002	.03	.02	.02	.02	.03	330
(T13Al-2V)																		
P144								.002	.01	.005	.005	.002	.005	.002	.02	.02	.02	255
P145	.060	.0040	.200	.031	.20	5.90	3.93	.002	.1	.005	.005	.002	.015	.002	.03	.02	.015	341
(T13V-4Fe)																		
P146	.032	.0022	.081	.016	.058	.01	14.00	.002	.015	.005	.005	.002	.015	.002	.015	.002	.002	293

TABLE J22 (Continued)

HEAT ANALYSES

Heat No.	C	H	O	N	Fe	Al	V	Cr	Co	Cu	Mg	Mn	NI	Pb	Si	Mo	Sn	BHN
(T17A1)-186-1V																		
(P147)	.012	.0027	.037	.0022	.0049	5.93	.98	.002	.1	.005	.005	.002	.005	.002	.02	3.00	.002	248
(T17A1)-186																		
(P148)	.022	.0023	.072	.0025	.0048	6.96	.01	.002	.014	.005	.005	.002	.005	.002	.02	4.10	.002	311
P153	.23	.0033	.17	.011	.15	6.12	3.95	.002	.015	.008	.005	.002	.015	.002	.02	.04	.003	300
P154	.087	.0025	.28	.033	.19	5.85	4.11	.002	.02	.01	.005	.003	.015	.002	.02	.02	.008	309
P155	.021	.0032	.16	.012	.14	6.05	4.09	.002	.02	.008	.005	.003	.02	.002	.025	.04	.002	308
P156	.057	.0034	.31	.046	.20	5.90	4.09	.002	.025	.005	.005	.002	.015	.002	.02	.04	.003	324
P157	.056	.0031	.22	.035	.19	5.80	3.93	.002	.04	.008	.005	.002	.03	.002	.01	.01	.004	317
P158	.041	.0028	.23	.024	.16	6.00	4.09	.002	.02	.02	.002	.003	.015	.005	.02	.02	.008	317
P159	.050	.0036	.29	.025	.17	6.00	3.55	.002	.02	.02	.002	.003	.015	.005	.02	.02	.008	311
P160	.037	.0028	.21	.020	.12	5.90	4.00	.002	.01	.005	.005	.002	.005	.002	.02	.01	.02	311
P161	.022	.0035	.16	.017	.14	6.00	3.96	.002	.015	.02	.005	.003	.02	.002	.03	.02	.05	306
P162	.089	.0028	.21	.040	.21	6.00	4.11	.002	.1	.008	.005	.003	.03	.002	.05	.03	.01	331
P163	.034	.0026	.14	.016	.13	5.95	3.99	.002	.015	.055	.055	.002	.005	.002	.1	.002	.015	306
P167	.015	.0044	.13	.016	.12	5.96	2.44	.002	.015	.005	.005	.003	.01	.002	.02	.015	.83	297
P168	.032	.0049	.19	.021	.14	6.05	4.04	.003	.015	.005	.005	.002	.01	.002	.02	.02	.03	311
P169	.041	.0031	.19	.019	.15	6.05	4.09	.002	.015	.005	.005	.002	.01	.002	.02	.01	.008	316
P170	No Pour - 111 Trade Broke																	
P171	.050	.0044	.19	.029	.17	5.95	4.00	.002	.02	.005	.005	.002	.01	.002	.02	.01	.05	331
P172	.060	.0046	.20	.029	.17	5.90	4.07	.002	.02	.005	.005	.003	.015	.002	.015	.015	.04	331
P173	.064	.0047	.29	.030	.23	5.90	4.08	.002	.025	.05	.005	.002	.02	.002	.02	.03	.02	311
P174	.093	.0041	.19	.022	.17	5.07	4.05	.002	.02	.015	.005	.002	.015	.002	.02	.015	.015	331
P175	.061	.0049	.21	.029	.17	5.87	4.07	.002	.02	.01	.005	.004	.03	.002	.02	.015	.008	326
P176	.060	.0049	.24	.030	.16	5.90	3.94	.002	.015	.005	.005	.002	.005	.002	.015	.03	.015	311
P177	.053	.0039	.25	.025	.15	6.10	4.03	.002	.015	.005	.005	.002	.009	.002	.02	.02	.04	331
P178	.074	.0041	.24	.034	.17	5.95	4.08	.002	.02	.005	.005	.002	.01	.002	.015	.02	.04	321
P179	.070	.0033	.29	.028	.16	5.95	3.53	.02	.02	.008	.005	.002	.015	.002	.02	.02	.04	336
P184	.049	.0039	.22	.027	.17	5.97	3.98	.002	.02	.005	.005	.002	.02	.002	.02	.03	.04	336
P185	.044	.0040	.29	.041	.17	6.05	3.98	.002	.02	.005	.005	.002	.015	.002	.02	.03	.04	341
P186	.044	.0045	.24	.021	.13	5.85	2.23	.002	.01	.005	.005	.002	.005	.002	.025	.01	.68	
P187	.036	.0038	.21	.026	.15	5.97	3.83	.002	.015	.005	.005	.002	.015	.002	.02	.02	.08	
P188	.030	.0041	.14	.019	.13	6.05	3.93	.002	.015	.005	.005	.004	.01	.002	.02	.05	.02	293
P191	.033	.0042	.29	.020	.14	5.18	3.14	.002	.01	.005	.005	.002	.005	.002	.015	.01	.004	
P194	.020	.0046	.24	.018	.13	6.05	4.00	.002	.01	.005	.005	.002	.005	.002	.02	.01	.002	
P195	.063	.0048	.30	.034	.16	5.75	3.83	.002	.015	.005	.005	.002	.01	.002	.02	.015	.03	336
P196	.024	.0059	.17	.017	.13	5.93	2.33	.002	.015	.005	.005	.002	.005	.002	.02	.005	.68	311
P197	.033	.0041	.14	.013	.13	5.90	3.30	.002	.01	.005	.005	.003	.005	.002	.015	.01	.1	285
P198	.013	.0042	.15	.022	.13	6.00	3.89	.002	.015	.005	.005	.003	.005	.002	.02	.02	.06	321
P199	.049	.0049	.15	.021	.13	5.87	3.97	.002	.01	.015	.005	.003	.01	.002	.02	.02	.01	
P200	.035	.0042	.25	.012	.16	5.95	3.92	.002	.02	.008	.005	.002	.015	.002	.02	.015	.05	336
P201	.049	.0041	.18	.018	.14	6.07	4.00	.002	.015	.008	.005	.002	.015	.002	.02	.02	.02	
P205	.035	.0047	.14	.015	.13	6.10	4.09	.004	.02	.01	.005	.002	.015	.002	.1	.01	.006	
P207	.025	.0052	.15	.010	.14	5.82	3.40	.002	.015	.005	.005	.003	.005	.002	.02	.03	.1	306
P208	.029	.0039	.11	.014	.13	6.05	5.15	.002	.015	.005	.005	.002	.005	.002	.02	.04	.01	311
P209	.044	.0032	.22	.028	.15	5.70	3.65	.002	.02	.02	.005	.003	.04	.002	.02	.03	.05	
P210	.064	.0029	.21	.029	.15	5.87	3.84	.002	.015	.005	.005	.002	.015	.002	.02	.02	.08	311
P211	.036	.0035	.13	.021	.14	5.90	3.92	.002	.02	.05	.005	.002	.015	.002	.03	.02	.015	305
P212	.042	.0035	.20	.031	.15	5.70	3.65	.002	.02	.05	.005	.002	.015	.002	.02	.03	.38	331
P213	.037	.0046	.17	.024	.15	5.82	3.30	.002	.015	.005	.005	.003	.01	.002	.025	.04	.05	326
P214	.044	.0029	.14	.020	.17	5.95	4.09	.002	.015	.015	.005	.003	.02	.002	.025	.02	.02	311
P215	.046	.0040	.15	.014	.13	6.05	4.03	.002	.015	.005	.005	.004	.005	.002	.015	.03	.02	311
P216	.041	.0034	.19	.045	.21	5.80	3.74	.002	.02	.03	.005	.002	.015	.002	.025	.07	.1	306
P217	.041	.0033	.21	.023	.16	6.00	4.09	.002	.015	.005	.005	.002	.01	.002	.02	.02	.05	296

TABLE J22 (Continued)

HEAT ANALYSES

Heat No.	C	H	O	N	Fe	Al	V	Cu	C	Cu	Mg	Mn	Si	Pb	Sl	Mo	Sn	BHN
P218	.043	.0048	.22	.029	.16	5.67	3.93	.002	.015	.005	.005	.003	.015	.002	.05	.015	.06	314
P219	No Pour - Contaminated Scrap																	
P220	.032	.0051	.18	.020	.18	6.00	3.96	.002	.01	.02	.005	.002	.015	.002	.02	.02	.02	305
P221	.024	.0051	.14	.014	.14	5.82	4.08	.002	.01	.005	.005	.003	.02	.002	.03	.015	.005	299
P224	.055	.0012	.18	.025	.17	6.03	3.98	.002	.015	.02	.005	.002	.005	.002	.02	.01	.03	308
P225	.048	.0030	.16	.032	.19	6.00	3.90	.002	.02	.005	.005	.004	.015	.002	.02	.04	.08	311
P227	.048	.0036	.14	.024	.16	6.05	4.09	.002	.02	.07	.005	.002	.015	.002	.025	.02	.05	317
P228	No Pour - Electrode Broke																	
P229	.042	.0033	.20	.026	.16	5.97	3.92	.002	.02	.005	.005	.002	.005	.002	.015	.008	.03	317
P231	.040	.0043	.15	.030	.15	6.00	3.96	.002	.015	.005	.005	.004	.015	.002	.02	.03	.05	306
P232	.030	.0037	.12	.021	.14	6.05	3.98	.002	.015	.005	.005	.004	.015	.002	.02	.04	.06	311
P235	.063	.0031	.15	.024	.18	5.90	3.90	.002	.015	.015	.005	.002	.01	.002	.02	.03	.08	316
P237	No Pour - Electrode Broke																	
P238	.052	.0037	.26	.026	.22	5.80	3.96	.002	.015	.005	.005	.032	.02	.002	.02	.02	.06	324
P239	.035	.0035	.13	.019	.16	6.00	4.05	.002	.015	.005	.005	.002	.015	.002	.02	.02	.03	308
P240	.050	.0039	.21	.026	.16	5.90	3.94	.002	.015	.005	.005	.002	.01	.002	.02	.02	.03	308
P241	.041	.0049	.18	.021	.14	6.02	4.09	.002	.02	.005	.055	.002	.015	.002	.015	.02	.006	299
P242	.027	.0041	.18	.071	.14	6.03	4.00	.002	.015	.015	.005	.003	.01	.002	.02	.02	.05	296
P243	.019	.0037	.10	.013	.17	6.05	4.10	.002	.01	.005	.005	.002	.01	.002	.02	.015	.002	311
P244	.030	.0040	.14	.018	.16	6.05	4.07	.002	.01	.005	.005	.003	.03	.002	.025	.025	.015	299
P245	.028	.0014	.073	.0059	.12	5.75	4.09	.02	.02	.005	.005	.002	.03	.002	.02	.02	.008	269
P246	.027	.0031	.15	.018	.15	5.95	4.11	.002	.01	.005	.005	.0002	.01	.002	.08	.015	.015	311
P247	.040	.0031	.21	.022	.18	5.90	3.96	.002	.015	.02	.005	.002	.01	.002	.025	.02	.05	305
P248	.032	.0033	.19	.022	.15	6.02	2.98	.002	.015	.005	.005	.002	.01	.002	.02	.02	.03	308
P249	.032	.0014	.068	.0064	.13	5.70	4.09	.002	.02	.02	.005	.002	.02	.002	.02	.005	.005	274
P250	.031	.0035	.18	.024	.15	5.95	3.94	.002	.015	.005	.005	.002	.015	.002	.1	.02	.05	293
P251	.022	.0040	.11	.016	.15	6.17	4.13	.002	.01	.1	.005	.002	.015	.002	.03	.02	.002	299
P252	.035	.0016	.07	.0072	.13	5.77	4.09	.002	.02	.05	.005	.002	.01	.002	.015	.015	.004	277
P253	.040	.0038	.16	.027	.16	5.95	4.05	.004	.02	.02	.005	.002	.01	.002	.02	.02	.015	311
P254	.035	.0031	.14	.019	.14	5.95	3.90	.002	.01	.005	.005	.002	.01	.002	.015	.015	.03	
P255	.046	.0019	.07	.0073	.14	5.82	4.09	.002	.02	.008	.005	.002	.01	.002	.02	.015	.004	277
P256	.033	.0051	.20	.021	.15	6.05	3.97	.002	.01	.005	.005	.002	.005	.002	.02	.008	.03	306
P257	.050	.0032	.19	.026	.18	5.92	3.97	.002	.015	.005	.005	.002	.008	.002	.025	.008	.02	
P258	.024	.0012	.12	.016	.15	6.10	3.98	.002	.01	.005	.005	.003	.015	.002	.03	.02	.01	
P259	.049	.0020	.087	.0092	.13	5.80	4.02	.002	.02	.005	.005	.002	.02	.002	.02	.02	.008	279
P260	.045	.0050	.16	.023	.17	5.95	4.00	.002	.01	.005	.005	.002	.015	.002	.02	.02	.04	296
P261	.040	.0037	.07	.028	.15	5.92	3.90	.002	.015	.02	.005	.003	.01	.002	.015	.02	.015	311
P262	.030	.0033	.15	.021	.16	5.95	4.07	.002	.015	.005	.005	.002	.015	.002	.025	.02	.01	
P263	.051	.0029	.17	.024	.17	5.87	3.92	.002	.015	.005	.005	.002	.01	.002	.08	.01	.01	308
P264	.054	.0033	.22	.026	.18	6.00	4.17	.002	.1	.01	.005	.003	.01	.002	.015	.015	.015	326
P265	.037	.0033	.19	.025	.16	6.03	4.00	.002	.015	.008	.005	.002	.008	.002	.02	.01	.04	311
P266	.031	.0022	.15	.021	.14	6.10	4.01	.002	.01	.005	.005	.002	.005	.002	.02	.01	.015	
P267	.024	.0013	.066	.0072	.12	5.85	4.08	.002	.01	.005	.005	.003	.01	.002	.015	.02	.008	271
P268	.037	.0031	.16	.025	.16	6.03	4.00	.002	.01	.005	.005	.002	.005	.002	.02	.01	.04	311
P269	.042	.0026	.17	.026	.14	5.05	4.03	.002	.01	.005	.005	.002	.005	.002	.02	.02	.015	327
P270	.047	.0033	.15	.023	.16	5.95	3.90	.002	.015	.005	.005	.002	.005	.002	.02	.02	.02	
P271	.027	.0017	.075	.0086	.12	5.95	4.07	.002	.01	.005	.005	.002	.01	.002	.02	.015	.002	279
P272	.040	.0034	.14	.025	.14	6.01	3.98	.002	.01	.005	.005	.002	.008	.002	.015	.008	.02	344
P273	.047	.0035	.17	.024	.16	6.03	4.03	.002	.015	.005	.005	.002	.005	.002	.02	.015	.02	316
P274	.042	.0033	.20	.022	.17	6.03	4.09	.002	.01	.005	.005	.002	.005	.002	.015	.005	.015	311
P275								.002	.01	.005	.005	.002	.01	.002	.02	.02	.03	311
P276	.047	.0021	.20	.025	.15	5.97	3.97	.002	.015	.005	.005	.002	.008	.002	.02	.008	.03	326
P277	.044	.0026	.22	.029	.16	5.72	3.97	.002	.015	.01	.005	.002	.01	.002	.02	.02	.05	321
P278	.040	.0035	.20	.027	.20	5.95	4.11	.002	.06	.005	.005	.002	.02	.002	.02	.02	.02	326
P279	.037	.0035	.18	.025	.16	5.97	3.93	.002	.015	.005	.005	.002	.008	.002	.01	.015	.03	311

TABLE J22 (Continued)

HEAT ANALYSES

Heat No.	C	H	N	Fe	Al	V	Cu	Cr	Mn	Ni	Pb	Si	Mg	Sb	Pb			
P280	.037	.0022	.096	.010	.12	5.80	4.03	.002	.02	.005	.005	.002	.02	.002	.02	.01	.004	287
P281	.044	.0045	.18	.026	.17	6.00	4.00	.002	.015	.04	.005	.002	.008	.007	.015	.02	.03	316
P282	.048	.0025	.20	.012	.12	5.87	4.09	.002	.015	.005	.005	.002	.008	.002	.02	.006	.002	282
P283	.042	.0024	.20	.025	.16	5.97	4.03	.002	.01	.005	.005	.007	.005	.002	.02	.003	.015	336
P284	.044	.0038	.13	.023	.14	5.95	3.98	.002	.02	.005	.005	.007	.007	.002	.1	.005	.005	108
P285A	.034	.0036	.12	.014	.15	6.03	4.09	.002	.01	.008	.005	.002	.008	.002	.015	.015	.004	296
P285B	.013	.0047	.17	.015	.16	6.03	4.14	.002	.015	.005	.005	.002	.01	.002	.08	.005	.005	299
P286	.031	.0029	.17	.023	.14	6.05	4.07	.002	.015	.005	.005	.002	.005	.002	.02	.015	.015	308
P288	.050	.0036	.20	.027	.16	5.70	3.57	.002	.025	.005	.005	.002	.005	.002	.015	.005	.015	321
P289	.043	.0018	.11	.0053	.14	5.80	4.05	.002	.015	.005	.005	.002	.008	.002	.02	.008	.02	277
P290	.060	.0056	.20	.030	.16	5.75	3.92	.002	.015	.005	.005	.003	.01	.002	.02	.01	.06	317
P291	.033	.0032	.19	.022	.15	5.85	4.07	.002	.015	.005	.005	.002	.005	.002	.02	.015	.02	311
P292	.048	.0042	.20	.024	.16	5.94	3.98	.002	.015	.005	.005	.002	.008	.002	.02	.008	.015	321
P293	.048	.0030	.21	.025	.15	5.97	4.03	.002	.02	.005	.005	.002	.008	.002	.015	.005	.003	316
P294	.041	.0036	.19	.024	.17	6.00	3.91	.002	.015	.005	.005	.002	.008	.002	.015	.005	.02	321
P296	.051	.0036	.22	.026	.17	5.90	4.07	.002	.02	.005	.005	.002	.005	.002	.015	.02	.015	
P303	.057	.0019	.082	.015	.15	5.77	4.10	.002	.02	.005	.005	.002	.008	.002	.015	.01	.003	279
P304	.049	.0036	.23	.015	.17	5.90	4.03	.002	.015	.005	.005	.002	.005	.002	.02	.01	.015	321
P305	.050	.0011	.22	.029	.17	5.75	3.87	.002	.02	.005	.005	.002	.005	.002	.03	.01	.02	324
P306	.047	.0029	.21	.025	.16	5.95	4.05	.002	.02	.005	.005	.002	.005	.002	.03	.01	.015	308
P307	.048	.0045	.22	.026	.16	5.87	4.00	.002	.02	.005	.005	.003	.015	.002	.025	.015	.08	305
P308	.044	.0038	.19	.026	.16	5.80	4.07	.002	.02	.005	.005	.002	.008	.002	.02	.015	.03	317
P309	.063	.0033	.22	.027	.16	5.97	4.02	.002	.015	.005	.005	.003	.008	.002	.015	.015	.03	317
P310	.052	.0046	.20	.028	.17	6.04	3.95	.002	.015	.005	.005	.003	.01	.002	.02	.015	.05	316
P311	.041	.0033	.19	.023	.16	5.80	4.07	.002	.015	.005	.005	.003	.01	.002	.02	.015	.02	321
P312	.049	.0045	.18	.027	.16	5.85	4.03	.002	.015	.008	.005	.002	.008	.002	.015	.01	.015	321
P314	.053	.0026	.21	.026	.17	5.80	4.07	.002	.03	.005	.005	.003	.01	.002	.02	.02	.03	336
P315	.046	.0032	.22	.024	.18	5.97	3.98	.002	.02	.01	.005	.002	.008	.002	.02	.015	.015	324
P316	.050	.0029	.16	.025	.19	5.92	4.05	.002	.015	.005	.005	.003	.008	.002	.015	.01	.02	314
P317	.044	.0031	.20	.026	.17	5.70	4.11	.002	.05	.005	.005	.002	.008	.002	.02	.02	.03	336
P318	.057	.0048	.14	.027	.18	6.10	3.90	.002	.015	.005	.005	.002	.005	.002	.015	.015	.03	316
P319	.058	.0052	.19	.025	.17	5.95	4.07	.002	.015	.005	.005	.002	.005	.002	.015	.015	.04	316
P320	.049	.0040	.19	.030	.15	5.87	3.99	.002	.015	.005	.005	.003	.005	.002	.015	.015	.04	321
P321	.052	.0035	.21	.026	.16	5.90	4.00	.002	.015	.005	.005	.002	.005	.002	.02	.01	.02	311
P322	.053	.0044	.20	.028	.16	5.92	4.00	.002	.015	.005	.005	.003	.008	.002	.02	.015	.03	331
P323	.031	.0047	.20	.021	.15	5.97	4.09	.002	.015	.005	.005	.003	.008	.002	.015	.015	.015	311
P324	.051	.0043	.21	.027	.16	5.85	3.98	.002	.02	.005	.005	.002	.01	.002	.02	.015	.04	326
P325	.051	.0057	.21	.025	.17	5.90	4.03	.002	.02	.005	.005	.002	.01	.002	.08	.004	.004	326
P326	.048	.0035	.21	.026	.16	5.92	3.98	.002	.02	.005	.005	.002	.01	.002	.08	.003	.004	316
P327	.018	.0037	.18	.023	.16	6.02	4.05	.002	.015	.005	.005	.003	.005	.002	.02	.008	.03	321
P328	.027	.0034	.15	.016	.16	5.87	4.11	.002	.015	.005	.005	.002	.008	.002	.02	.01	.004	311
P329	.027	.0033	.15	.017	.14	6.05	4.02	.002	.015	.02	.005	.002	.01	.002	.02	.01	.02	311
P330	.049	.0036	.21	.024	.16	5.95	4.00	.002	.015	.008	.005	.002	.008	.002	.02	.015	.05	321
P332	.037	.0016	.065	.0024	.13	5.75	4.05	.002	.025	.005	.005	.002	.02	.002	.03	.004	.002	272
P333	.026	.0019	.14	.017	.16	6.08	4.06	.002	.015	.015	.005	.003	.005	.002	.015	.002	.01	302
P334	.040	.0042	.083	.011	.13	5.60	4.03	.002	.025	.005	.005	.002	.015	.002	.08	.004	.002	290
P335	.053	.0043	.22	.027	.16	5.92	4.05	.002	.04	.005	.005	.002	.008	.002	.02	.01	.02	325
P336	.046	.0032	.22	.024	.18	5.97	3.98	.002	.02	.005	.005	.002	.005	.002	.02	.01	.02	311
P337	.053	.0035	.20	.025	.16	5.95	4.03	.002	.02	.03	.005	.002	.005	.002	.015	.01	.02	
P340	.044	.0029	.19	.027	.16	5.85	4.02	.002	.02	.005	.005	.002	.005	.002	.02	.01	.02	327
P342	.060	.0036	.21	.029	.17	5.90	4.00	.002	.02	.005	.005	.002	.008	.002	.02	.015	.03	327
P343	.064	.0036	.24	.030	.17	5.87	4.07	.002	.01	.005	.005	.002	.005	.002	.02	.005	.02	324
P344	.055	.004	.20	.030	.17	5.90	4.07	.002	.02	.005	.005	.002	.005	.002	.02	.01	.02	343
P345	.044	.0031	.20	.026	.16	5.94	4.00	.002	.02	.01	.005	.002	.008	.002	.02	.01	.04	317
P346	.040	.0036	.20	.022	.15	5.92	4.07	.002	.01	.005	.005	.002	.005	.002	.015	.01	.015	311

TABLE J22 (Continued)

HEAT ANALYSES

Heat No.	C	H	O	N	Fe	Al	V	Co	Cr	Cu	Hg	Mn	NI	Pb	SI	Mo	Su	BPH
P342	.059	.0034	.20	.029	.17	5.90	4.04	.002	.02	.005	.005	.002	.005	.002	.015	.01	.02	331
P345	.048	.0016	.06	.028	.080	5.72	4.13	.002	.01	.005	.005	.002	.008	.002	.02	.005	.002	277
P350	.025	.0022	.06	.079	.075	5.80	4.11	.002	.01	.1	.005	.002	.005	.002	.01	.002	.002	290
P357	.020	.0020	.05	.045	.075	5.87	4.13	.002	.01	.005	.005	.002	.008	.002	.02	.005	.002	287
P358	.022	.0020	.22	.027	.17	5.80	4.07	.002	.02	.005	.005	.002	.005	.002	.02	.005	.015	327
P361	.075	.0026	.21	.032	.18	5.94	3.88	.002	.015	.01	.005	.003	.005	.002	.015	.22	.015	324
P362	.041	.0029	.71	.023	.15	5.67	3.94	.002	.015	.005	.005	.002	.005	.002	.015	.02	.008	314
P363	.052	.0041	.24	.027	.15	5.97	3.97	.002	.015	.008	.005	.002	.005	.002	.02	.015	.03	317
P364	.045	.0040	.24	.025	.15	5.60	3.83	.002	.015	.01	.005	.002	.005	.002	.02	.015	.01	305
P365	.073	.0041	.24	.031	.16	6.15	3.97	.002	.02	.008	.005	.002	.005	.002	.02	.01	.02	311
P366	.051	.0033	.24	.024	.16	6.25	4.03	.002	.015	.005	.005	.003	.005	.002	.02	.005	.02	311
P367	.058	.0032	.18	.029	.16	6.00	4.05	.002	.015	.01	.005	.003	.005	.002	.02	.005	.02	346
P368	.083	.0052	.24	.030	.18	5.90	4.04	.002	.025	.01	.005	.002	.005	.002	.015	.015	.02	327
P369	.063	.0036	.23	.027	.16	5.90	4.05	.002	.02	.01	.005	.002	.005	.002	.015	.015	.02	317
P370	.073	.0033	.24	.031	.18	5.90	4.06	.002	.015	.01	.005	.003	.005	.002	.015	.02	.02	314
P371	.063	.0040	.27	.048	.18	6.06	3.98	.02	.02	.005	.005	.002	.01	.002	.1	.005	.008	331
P372	.049	.0026	.11	.012	.14	5.72	4.02	.002	.015	.008	.005	.002	.01	.002	.02	.005	.004	279
P373	.064	.0026	.24	.030	.17	5.95	4.00	.002	.015	.008	.005	.003	.005	.002	.02	.005	.02	324
P374	.064	.0028	.25	.030	.21	5.95	4.03	.002	.02	.01	.005	.003	.005	.002	.015	.002	.03	317
P375	.042	.0033	.12	.010	.13	5.80	4.05	.002	.025	.008	.005	.002	.01	.002	.025	.005	.004	287
P376	.066	.0028	.14	.016	.14	5.70	4.05	.002	.02	.008	.005	.002	.01	.002	.025	.005	.004	296
P378	.048	.0033	.26	.023	.15	6.05	3.98	.002	.025	.005	.005	.002	.01	.002	.1	.005	.008	317.6
P379	.039	.0034	.26	.038	.22	5.85	3.93	.002	.01	.008	.005	.003	.005	.002	.015	.004	.02	334.3
P380	.073	.0020	.26	.033	.20	5.90	3.98	.002	.01	.005	.005	.003	.005	.002	.015	.02	.015	331
P381	.085	.0053	.25	.030	.20	6.10	4.07	.002	.03	.005	.005	.003	.03	.002	.1	.01	.008	331
P382	.033	.0031	.27	.031	.19	6.00	4.04	.002	.02	.005	.005	.002	.02	.002	.1	.01	.008	331
P383	.066	.0027	.26	.031	.17	6.04	3.87	.002	.01	.01	.005	.002	.005	.002	.015	.004	.02	316
P384	.057	.0058	.24	.028	.16	6.03	3.91	.002	.01	.005	.005	.002	.005	.002	.015	.015	.015	326
P385	.066	.0027	.24	.032	.20	6.05	3.81	.002	.015	.008	.005	.003	.005	.002	.015	.01	.02	311
P386	.078	.0049	.23	.035	.20	5.95	4.00	.002	.01	.005	.005	.002	.005	.002	.015	.015	.015	306.5
P387	.074	.0036	.26	.033	.19	6.02	3.97	.002	.02	.008	.005	.003	.005	.002	.01	.01	.02	331
P388	.076	.0041	.25	.031	.20	5.85	3.98	.002	.01	.008	.005	.003	.005	.002	.01	.015	.015	326
P389	.080	.0041	.27	.032	.20	5.80	4.00	.002	.015	.008	.005	.002	.005	.002	.015	.01	.015	346.5
P390	.074	.0046	.28	.041	.19	5.96	3.75	.002	.015	.008	.005	.002	.005	.002	.015	.015	.02	341.5
P391	.078	.0045	.26	.032	.21	5.85	4.00	.002	.015	.01	.005	.002	.005	.002	.015	.01	.02	326
P392	.051	.0028	.24	.028	.17	6.00	4.04	.002	.04	.005	.005	.002	.02	.002	.1	.01	.01	346.5
P393	.065	.0030	.25	.030	.19	5.95	3.88	.002	.02	.008	.005	.002	.005	.002	.02	.01	.015	352
P394	.046	.0021	.22	.025	.18	6.15	4.04	.002	.03	.008	.005	.003	.01	.002	.03	.01	.01	331
P395	.053	.0027	.22	.025	.16	6.25	4.03	.002	.01	.005	.005	.003	.005	.002	.01	.01	.02	
P396	.053	.0026	.22	.026	.16	6.05	4.05	.002	.015	.005	.005	.002	.005	.002	.015	.01	.02	
P397	.057	.0026	.22	.024	.17	6.20	4.02	.002	.02	.005	.005	.003	.008	.002	.015	.01	.02	
P398	.067	.0028	.23	.029	.20	6.02	3.91	.002	.02	.01	.005	.002	.005	.002	.01	.02	.015	323
P399	.025	.0049	.32	.031	.17	6.20	4.07	.002	.015	.005	.005	.004	.015	.002	.01	.01	.02	302
P400	.046	.0034	.24	.030	.16	6.10	3.95	.002	.025	.005	.005	.002	.01	.002	.1	.005	.002	346
P401	.046	.0036	.21	.029	.16	6.11	4.00	.002	.015	.008	.005	.003	.015	.002	.015	.01	.015	311
P402	.068	.0023	.23	.031	.16	6.05	4.05	.003	.025	.02	.005	.002	.02	.002	.1	.015	.008	326
P403	.060	.0027	.26	.033	.16	6.05	4.01	.002	.02	.005	.005	.002	.01	.002	.08	.005	.01	321
P404	.052	.0023	.22	.029	.16	6.05	4.04	.002	.03	.008	.005	.002	.01	.002	.05	.005	.05	331.3
P405	.069	.0039	.25	.030	.17	5.75	4.01	.002	.03	.005	.005	.004	.005	.002	.015	.02	.03	337.6
P406	.064	.0045	.25	.030	.16	5.85	4.01	.002	.04	.005	.005	.002	.01	.002	.1	.005	.004	347.6
P410	.062	.0027	.24	.027	.16	6.05	4.05	.002	.03	.005	.005	.002	.005	.002	.025	.01	.008	327.5
P411	.062	.0033	.23	.028	.16	5.95	4.07	.002	.015	.008	.005	.002	.005	.002	.025	.01	.008	324.4
P412	.054	.0026	.24	.026	.16	5.85	4.02	.002	.015	.005	.005	.002	.005	.002	.025	.005	.004	331
P413	.077	.0033	.17	.019	.14	6.10	4.09	.002	.02	.005	.005	.002	.01	.002	.1	.002	.002	314.3
P414	.066	.0032	.23	.030	.13	6.05	3.95	.002	.015	.01	.005	.003	.01	.002	.01	.015	.015	317.5

TABLE J22 (Continued)

HEAT ANALYSES

Heat No.	C	H	O	N	Fe	Al	V	Cu	Cr	Co	Mg	Mn	Ni	Pb	Si	Mo	Sr	BHN
P415	.045	.0075	.26	.023	.15	5.95	4.08	.002	.02	.005	.005	.002	.01	.002	.02	.002	.002	
P416	.058	.0021	.26	.050	.21	6.00	4.01	.002	.05	.005	.005	.002	.005	.002	.02	.03	.015	316
P417					No Pump - Hot Water													
P418	.041	.0026	.19	.019	.17	6.10	4.09	.002	.115	.01	.005	.002	.01	.002	.03	.008	.015	321
P419	.064	.0022	.26	.036	.17	6.00	4.06	.002	.02	.005	.005	.002	.005	.002	.025	.006	.015	337.5
P420	.038	.0021	.19	.015	.16	6.10	4.13	.002	.02	.005	.005	.002	.01	.002	.03	.006	.008	321
P421	.029	.0025	.17	.014	.13	6.20	4.05	.002	.02	.005	.005	.002	.01	.002	.1	.005	.002	
P422	.066	.0030	.25	.022	.17	6.15	4.05	.002	.02	.005	.005	.003	.01	.002	.015	.015	.02	315.3
P423	.056	.0030	.21	.021	.14	6.15	4.05	.002	.025	.005	.005	.002	.005	.002	.015	.02	.02	308
P424	.043	.0026	.25	.017	.15	6.10	4.05	.002	.015	.01	.005	.004	.005	.002	.015	.015	.008	314.3
P425	.043	.0023	.21	.019	.16	6.15	4.10	.002	.02	.005	.005	.003	.015	.002	.1	.005	.004	321.6
P426	.024	.0033	.17	.015	.13	6.20	4.00	.002	.02	.005	.005	.002	.005	.002	.03	.008	.002	308
P427	.033	.0054	.16	.019	.15	6.25	4.03	.002	.002	.005	.005	.002	.01	.002	.02	.008	.002	311.3
P428	.043	.0025	.17	.014	.11	6.20	4.10	.002	.015	.005	.005	.002	.005	.002	.05	.003	.002	297.5
P429	.062	.0022	.23	.028	.17	6.00	4.04	.002	.06	.005	.005	.002	.05	.002	.03	.002	.002	
P430	.077	.0026	.12	.012	.15	5.90	4.09	.002	.01	.02	.005	.004	.005	.002	.015	.003	.003	311.3
P441	.050	.0019	.16	.023	.16	6.10	4.03	.002	.025	.005	.005	.002	.02	.002	.10	.003	.002	314.3
P442	.057	.0022	.25	.010	.17	6.10	3.97	.002	.01	.005	.005	.003	.005	.002	.01	.02	.015	314.2
P443	.039	.0021	.19	.021	.15	6.15	4.09	.002	.02	.02	.005	.002	.01	.002	.08	.005	.003	
P444	.064	.0024	.19	.028														
P445	.050	.0024	.18	.020	.16	6.15	4.05	.002	.01	.008	.005	.008	.01	.002	.015	.02	.015	316
P446	.033	.0041	.17	.017	.16	6.15	4.05	.002	.015	.015	.005	.004	.01	.002	.015	.015	.01	308
P447	.021	.0034	.17	.016	.14	6.15	4.06	.002	.02	.005	.005	.002	.02	.002	.16	.002	.002	311
P448	.032	.0031	.19	.015														
P449	.042	.0024	.18	.017	.19	5.90	3.96	.002	.025	.01	.005	.002	.015	.002	.1	.002	.002	302
P450	.050	.0025	.23	.029	.16	6.05	4.07	.002	.02	.005	.005	.002	.01	.002	.05	.002	.002	314.3
P452	.050	.0025	.21	.020	.16	6.15	4.07	.005	.015	.03	.005	.003	.005	.002	.015	.01	.015	314.3
P453	.041	.0023	.23	.025	.16	5.90	3.95	.002	.015	.01	.005	.002	.005	.002	.015	.004	.015	321
P454	.049	.0026	.17	.021	.15	6.15	4.03	.002	.015	.008	.005	.002	.005	.002	.01	.004	.008	311
P455	.044	.0026	.20	.022	.15	6.00	4.04	.002	.015	.005	.005	.002	.005	.002	.04	.002	.002	317.6
P456	.073	.0022	.19	.023	.16	6.00	4.05	.002	.01	.005	.005	.002	.005	.002	.01	.002	.004	327.6
P457	.051	.0036	.15	.021	.16	6.15	4.09	.002	.015	.008	.005	.01	.01	.002	.015	.005	.006	
P458	.060	.0023	.23	.024	.15	6.10	4.04	.002	.02	.005	.005	.002	.01	.002	.08	.005	.004	331
P459	.040	.0022	.20	.020	.16	6.15	4.04	.002	.015	.008	.005	.002	.01	.002	.015	.005	.002	321
P460	.037	.0034	.20	.019	.17	6.00	4.04	.002	.02	.005	.005	.002	.01	.002	.08	.002	.002	311
P461	.062	.0033	.23	.022	.16	5.95	4.09	.002	.015	.005	.005	.003	.005	.002	.02	.002	.004	
P462	.021	.0024	.21	.024	.16	5.90	4.05	.002	.02	.005	.005	.002	.005	.002	.015	.002	.004	
P463	.050	.0034	.22	.026	.16	6.00	4.08	.002	.01	.005	.005	.002	.005	.002	.01	.002	.01	
P464	.042	.0031	.21	.022	.16	6.10	4.04	.002	.01	.005	.005	.003	.005	.002	.015	.002	.003	
P465	.023	.0028	.21	.021	.15	6.00	4.05	.002	.01	.005	.005	.002	.005	.002	.02	.002	.003	
P466	.044	.0023	.16	.026	.17	6.10	4.06	.002	.015	.005	.005	.008	.015	.002	.02	.004	.004	321
P467	.028	.0023	.20	.027	.16	5.95	4.03	.002	.01	.005	.005	.003	.005	.002	.01	.01	.008	
P468	.070	.0026	.12	.022	.18	6.00	4.04	.002	.01	.005	.005	.003	.005	.002	.01	.01	.008	
P469	.042	.0021	.19	.024	.15	5.90	4.05	.002	.015	.005	.005	.004	.005	.002	.015	.01	.01	
P470	.048	.0022	.18	.022	.18	5.90	4.04	.002	.01	.005	.005	.004	.008	.002	.015	.003	.008	
P473	.062	.0021	.24	.025	.15	5.90	4.09	.002	.015	.005	.005	.002	.005	.002	.015	.002	.01	316.3
P474	.024	.0026	.23	.025	.17	6.05	4.07	.002	.015	.015	.005	.003	.01	.002	.01	.004	.01	324.3
P475	.026	.0022	.18	.022	.15	5.95	4.07	.002	.01	.005	.005	.003	.005	.002	.01	.003	.004	408
P476	.054	.0023	.16	.024	.18	6.05	4.07	.002	.01	.005	.005	.002	.005	.002	.01	.002	.005	321
P477	.051	.0025	.22	.024	.17	5.90	4.05	.002	.01	.005	.005	.002	.005	.002	.01	.003	.002	321
P478	.045	.0015	.20	.024	.17	6.01	4.1	.002	.02	.005	.005	.002	.005	.002	.01	.002	.005	
P479	.015	.0019	.19	.019	.15	6.00	4.05	.002	.01	.005	.005	.002	.005	.002	.01	.01	.002	316
P481	.053	.0027	.19	.024	.15	5.90	4.01	.002	.01	.005	.005	.003	.005	.002	.01	.01	.01	316.3
P482	.043	.0023	.23	.026	.17	6.00	4.04	.002	.01	.005	.005	.003	.005	.002	.01	.002	.002	
P483	.026	.0023	.19	.021	.15	6.00	4.05	.002	.01	.005	.005	.002	.005	.002	.01	.002	.002	

TABLE J22 (Continued)

HEAT ANALYSES

Heat No.	C	H	N	O	Fe	Al	S	Si	Ca	Co	Pb	Zn	Mn	Mo	Se	As	Sb	Bism
P485	.051	.0024	.22	.028	.17	5.75	5.05	.002	.015	.00	.00	.003	.01	.002	.015	.009	.006	331
P486	.054	.0023	.19	.026	.16	6.00	5.04	.002	.01	.00	.00	.003	.005	.002	.01	.015	.006	329.3
P487	.058	.0020	.27	.027	.16	6.05	5.05	.002	.015	.00	.00	.004	.005	.002	.015	.002	.003	352
P490	.030	.0025	.21	.022	.15	6.10	5.07	.002	.015	.00	.00	.015	.005	.002	.021	.005	.009	343
P494	.064	.0026	.15	.023	.17	6.10	5.07	.002	.015	.01	.00	.003	.006	.002	.015	.01	.01	316
P495	.057	.003	.191	.027	.17	6.05	5.09	.002	.015	.00	.00	.003	.005	.002	.01	.005	.01	321
P496	.061	.0029	.19	.026	.17	6.10	5.04	.002	.01	.005	.00	.002	.005	.002	.01	.003	.005	327.5
P497	.056	.0030	.22	.027	.16	6.10	5.04	.002	.01	.01	.00	.004	.01	.002	.01	.004	.015	327.6
P498	.030	.002	.186	.024	.16	6.00	5.08	.002	.01	.00	.00	.003	.005	.002	.01	.004	.006	317.6
P499	.053	.0026	.20	.023	.17	6.06	5.09	.002	.015	.01	.00	.003	.005	.002	.01	.005	.01	324.3
P500	.095	.0023	.16	.021	.16	6.00	5.05	.002	.01	.015	.00	.004	.005	.002	.01	.004	.006	324.3
P502	.076	.0021	.20	.026	.16	5.95	5.05	.002	.01	.005	.005	.004	.015	.002	.015	.00	.008	341
P503	.045	.0036	.15	.025	.18	6.05	5.05	.002	.015	.006	.00	.003	.008	.002	.021	.015	.01	311
P504	.052	.0027	.12	.023	.16	6.12	5.07	.002	.015	.005	.00	.006	.005	.002	.015	.01	.008	331
P505	.045	.0029	.20	.022	.17	6.00	5.03	.002	.01	.01	.00	.004	.005	.002	.01	.01	.015	326
P506	.068	.002	.216	.025	.17	6.10	5.05	.002	.015	.005	.00	.003	.005	.002	.01	.008	.01	331
P507	.054	.002	.195	.022	.17	5.55	5.13	.002	.015	.004	.00	.004	.006	.002	.01	.004	.005	326
P508	.052	.003	.177	.025	.18	6.05	5.06	.002	.015	.006	.00	.003	.008	.00	.01	.002	.003	336
P509	.055	.002	.21	.025	.17	6.10	5.11	.002	.015	.01	.00	.003	.005	.002	.015	.005	.006	311
P510	.055	.001	.124	.023	.14	6.05	5.09	.002	.01	.005	.005	.002	.005	.002	.01	.002	.003	331
P511	.056	.003	.151	.011	.16	6.00	5.06	.002	.01	.005	.00	.004	.006	.002	.01	.002	.003	311
P512	.046	.0030	.20	.024	.15	6.15	5.03	.002	.01	.006	.00	.003	.005	.002	.015	.01	.01	325.3
P524	.046	.0030	.21	.029	.16	6.10	5.05	.002	.01	.01	.00	.003	.005	.002	.01	.004	.009	331
P525	.051	.006	.161	.03	.17	6.00	5.05	.002	.015	.005	.00	.003	.005	.002	.015	.005	.01	321
P526	.055	.0035	.21	.025	.16	6.00	5.07	.002	.01	.005	.005	.003	.005	.002	.01	.002	.01	331
P528	.025	.0050	.22	.016	.16	6.50	5.12	.002	.01	.005	.00	.004	.005	.002	.01	.002	.002	311.3
P529	.046	.0030	.22	.025	.17	6.20	5.07	.002	.01	.005	.00	.004	.005	.002	.01	.003	.008	321
P530	.055	.0023	.25	.026	.16	6.05	5.05	.002	.01	.01	.00	.003	.005	.002	.01	.003	.006	324
P531	.050	.0030	.24	.020	.17	6.15	5.04	.002	.01	.005	.00	.002	.005	.002	.01	.003	.006	317.6
P532	.053	.0026	.23	.024	.15	6.05	5.06	.002	.015	.01	.00	.004	.01	.002	.01	.003	.008	327.3
P533	.051	.0040	.21	.027	.17	6.10	5.05	.002	.015	.00	.00	.003	.005	.002	.01	.01	.01	324.3
P534	.053	.0050	.21	.026	.19	6.05	5.07	.003	.02	.00	.00	.003	.005	.002	.01	.01	.01	331
P535	.055	.0032	.21	.024	.17	6.00	5.04	.002	.01	.006	.005	.004	.005	.002	.01	.01	.01	331
P536	.053	.0039	.22	.024	.16	6.00	5.09	.002	.01	.005	.00	.003	.01	.002	.01	.006	.004	
P537	.050	.0029	.21	.024	.15	6.05	5.05	.002	.01	.01	.00	.004	.01	.002	.01	.002	.006	
P538	.051	.0026	.21	.022	.16	6.05	5.11	.002	.01	.005	.005	.003	.01	.002	.01	.004	.008	
P539	.067	.0027	.23	.024	.16	5.90	5.05	.002	.015	.002	.005	.005	.01	.002	.01	.006	.01	321
P540	.047	.0050	.22	.025	.18	6.00	5.08	.002	.01	.00	.00	.004	.005	.002	.01	.008	.01	314.4
P541	.057	.0030	.23	.031	.16	5.05	5.05	.002	.01	.00	.00	.004	.006	.002	.015	.015	.015	321
P542	No Record																	335
P543	.063	.0028	.2	.025	.16	6.05	5.04	.002	.02	.003	.005	.005	.01	.002	.015	.01	.006	327.3
P544	.067	.0023	.16	.016	.15	6.05	5.07	.002	.01	.01	.00	.003	.005	.002	.015	.01	.01	
P545	.069	.0025	.16	.025	.16	5.70	5.05	.002	.01	.005	.00	.002	.005	.002	.01	.005	.009	
P546	.060	.0024	.19	.029	.15	6.1	5.12	.00	.01	.005	.00	.003	.01	.002	.01	.004	.004	326.5
P547	No Record																	
P548	.062	.0026	.22	.025	.16	6.05	5.04	.002	.015	.00	.00	.004	.01	.002	.015	.005	.006	312
P549	.055	.0025	.22	.024	.15	6.00	5.07	.002	.015	.005	.00	.004	.005	.002	.01	.003	.002	311
P550	.056	.0025	.20	.024	.15	6.05	5.07	.00	.02	.00	.00	.004	.005	.002	.01	.003	.002	311
P552	.059	.0025	.23	.025	.15	6.00	5.05	.002	.015	.00	.00	.003	.005	.002	.01	.005	.01	321
P553	.052	.0025	.23	.026	.16	6.00	5.05	.002	.01	.01	.00	.003	.005	.002	.01	.005	.01	
P554	.056	.0031	.23	.025	.15	6.05	5.04	.002	.01	.00	.00	.003	.005	.002	.01	.005	.003	315
P555	.053	.0027	.15	.024	.15	6.05	5.04	.00	.01	.00	.00	.003	.005	.002	.01	.005	.003	312
P556	.054	.0025	.15	.024	.15	6.10	5.04	.00	.01	.00	.00	.003	.005	.002	.01	.005	.003	312
P557	.054	.0025	.15	.024	.15	6.05	5.04	.00	.01	.00	.00	.003	.005	.002	.01	.005	.003	312
P558	.054	.0025	.15	.024	.15	6.10	5.04	.00	.01	.00	.00	.003	.005	.002	.01	.005	.003	312

TABLE J22 (Continued)

HEAT ANALYSES

Heat	C	H	a	H	Fe	Al	v	Si	U	Co	Mg	Mn	Ni	Cr	Si	Mo	Sr	Others	
P559	.025	.0025	.20	.036	.13	5.75	4.11	.002	.015	.05	.005	.006	.005	.002	.015	.003	.006	30.3	
P560	.032	.0025	.22	.039	.22	6.13	5.77	.002	.015	.008	.005	.006	.006	.002	.01	.01	.006	33.1	
P561	.031	.0027	.19	.030	.12	5.10	4.05	.002	.02	.008	.005	.002	.01	.002	.02	.01	.005	29.5	
P562	.026	.003	.25	.052	.14	5.90	4.00	.002	.015	.003	.005	.006	.031	.002	.02	.02	.006	31.1	
P563	.023	.0025	.22	.032	.17	5.95	4.05	.002	.015	.006	.005	.006	.005	.002	.01	.005	.008		
P564	.061	.0026	.21	.033	.17	5.93	4.05	.002	.02	.015	.005	.007	.01	.002	.015	.01	.01	32.1	
P565	.015	.0027	.19	.015	.25	5.95	4.13	.001	.01	.01	.005	.004	.01	.002	.01	.01	.01	31.1	
P566	.012	.0027	.17	.042	.12	6.05	4.10	.002	.01	.005	.005	.006	.015	.002	.01	.015	.008		
P567	No Record																		
P568	.052	.0026	.24	.041	.15	6.00	4.09	.002	.015	.005	.005	.006	.015	.002	.015	.015	.006		
P569	.065	.0028	.25	.037	.15	5.90	4.12	.002	.015	.005	.005	.006	.01	.002	.01	.005	.008	30.2	
P570	.075	.0028	.21	.041	.15	5.95	4.02	.002	.01	.005	.005	.004	.01	.002	.01	.004	.006		
P571	.072	.0023	.26	.061	.14	5.90	4.07	.002	.015	.005	.005	.006	.01	.002	.015	.004	.006	31.1	
P572	.063	.0025	.22	.037	.14	6.02	4.06	.002	.015	.008	.005	.006	.01	.002	.0152	.08	.005	31.1	
P573	.062	.0026	.19	.050	.16	5.95	4.08	.002	.015	.008	.005	.006	.01	.002	.01	.004	.008	31.1	
P574	.008	.0024	.21	.037	.16	6.02	4.07	.002	.015	.015	.005	.004	.01	.002	.01	.004	.006	31.1	
P575	.010	.0032	.26	.052	.16	5.97	5.04	.002	.015	.008	.005	.003	.006	.002	.015	.02	.006	30.1	
P576	.021	.0023	.25	.033	.16	6.02	4.06	.002	.015	.006	.005	.004	.01	.002	.01	.004	.01	32.1	
P577	.002	.0026	.17	.015	.17	6.02	4.09	.002	.015	.01	.005	.006	.002	.01	.002	.01	.005	.008	30.2
P578	.012	.0033	.16	.017	.13	5.97	4.13	.002	.01	.005	.005	.003	.005	.002	.02	.02	.006	31.1	
P579	.055	.0028	.22	.047	.11	.005	5.00	.05	.01	.005	.005	.002	.002	.002	.01	.002	.006		
P580	.036	.0031	.18	.057	.15	5.77	5.03	.002	.01	.005	.005	.006	.003	.006	.01	.01	.006	31.1	
P581	.023	.0033	.18	.013	.16	6.02	5.03	.002	.01	.005	.005	.003	.003	.002	.01	.01	.003	30.2	
P582	.021	.0036	.15	.017	.15	6.02	4.11	.002	.01	.005	.005	.004	.003	.002	.01	.01	.005	31.1	
P583	No Print																		
P584	.017	.0037	.15	.017	.15	5.05	4.11	.002	.01	.005	.005	.002	.005	.002	.03	.015	.002	30.2	
P585	.062	.0031	.23	.065	.15	5.98	4.10	.002	.02	.01	.005	.002	.015	.006	.01	.002	.002	31.1	
P586	.035	.0030	.26	.063	.13	5.77	4.15	.002	.015	.015	.005	.002	.008	.002	.03	.015	.002	30.2	
P587	.063	.0026	.23	.031	.05	6.12	4.06	.002	.02	.01	.005	.002	.015	.006	.01	.002	.002	32.1	
P588	.029	.0039	.20	.026	.16	6.05	4.11	.002	.015	.005	.005	.002	.006	.002	.03	.015	.002	31.1	
P589	.030	.0033	.19	.023	.13	6.05	4.09	.002	.015	.005	.005	.002	.008	.002	.025	.015	.002	31.1	
P590	.042	.0033	.21	.033	.16	6.10	4.06	.002	.015	.005	.005	.002	.005	.002	.02	.015	.003	31.1	
P591	.065	.0030	.25	.033	.13	5.90	4.06	.002	.02	.006	.005	.002	.015	.002	.02	.003	.002	31.1	
P592	.029	.0036	.24	.046	.16	5.97	4.09	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P593	.070	.0025	.22	.031	.15	6.20	4.08	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P594	.050	.0028	.23	.029	.14	5.60	4.09	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P595	.042	.0027	.23	.032	.16	6.15	4.09	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P596	.003	.0027	.23	.036	.05	6.15	4.05	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P599	.029	.0030	.24	.032	.12	6.00	4.03	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P600	.003	.0027	.26	.035	.16	5.75	4.03	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P601	.016	.0033	.17	.015	.13	5.96	4.11	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P602	.003	.0030	.19	.035	.16	5.77	4.08	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P603	.016	.0033	.13	.014	.14	6.03	4.05	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P604	.033	.0031	.17	.019	.13	6.17	4.06	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P605	.032	.0030	.15	.017	.16	6.13	4.07	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P606	.014	.0028	.20	.015	.05	6.03	4.07	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P609	.051	.0031	.25	.031	.17	6.09	4.09	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P616	.021	.0025	.25	.019	.14	6.03	4.13	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P611	.016	.0027	.20	.017	.15	6.10	4.05	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P612	.022	.0025	.17	.017	.14	6.05	4.05	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	31.1	
P613	.022	.0025	.17	.017	.14	6.05	4.05	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	30.2	
P614	.022	.0025	.17	.017	.14	6.05	4.05	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	30.2	
P615	.033	.0025	.21	.015	.05	6.03	4.07	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05	30.2	
P617	.020	.0021	.23	.013	.17	6.20	4.05	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05		
P618	.023	.0025	.20	.015	.05	6.03	4.07	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05		
P619	.023	.0023	.16	.01	.12	6.05	4.07	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05		
P620	.023	.0023	.16	.01	.12	6.05	4.07	.002	.05	.05	.05	.05	.05	.05	.05	.05	.05		

TABLE J23

RESULTS OF DEVELOPMENTAL BRACKET COMPARATIVE FATIGUE TESTS

Load, Pounds	Fatigue Life, Cycles	
	Cast Ti-5Al-4V (As-Cast)	Cast 410 Steel (180,000 psi, min.)
18,000	NT	21,000 (2)
15,000	5,010 (1)	26,700 (2)
15,000	2,542 (1)	30,380 (2)
12,000	7,552 (1)	93,180 (3)
12,000	14,710 (1)	118,450 (2)
9,000	30,000 (1)	430,330 (1)
9,000	34,149 (2)	311,000 (2)
7,000	140,051 (3)	NT
7,000	134,420 (2)	NT
5,000	448,618 (1)	NT
5,000	318,550 (2)	NT

NT - None Tested.

(1) Failure in upper flange.

(2) Failure in upper large lug, at mounting hole.

(3) Failure in small lug.

TABLE J24

RESULTS OF STATISTICAL EVALUATION OF DIMENSIONS

Dimension Type -- See Footnotes	Specified Nominal Dimension	Actual Average Dimension	Specified Range	Actual Range*
1	.200	0.205	$\pm .015$	$\pm .011$
1	.650	.674	$\pm .015$	$\pm .015$
1	3.050	3.099	$\pm .015$	$\pm .019$
1	Not Specified	9.598	$\pm .050$	$\pm .050$
2	1.520	1.533	$\pm .025$ $- .015$	$\pm .028$
2	2.520	2.525	$\pm .025$ $- .015$	$\pm .028$
3	Not Specified	.664	$\pm .015$	$\pm .035$
3	3.050	3.110	$\pm .015$	$\pm .037$
3	10.630	10.703	$\pm .050$	$\pm .071$

* Actual range was from conventional statistical analysis and represents plus and minus three standard deviations.

1. Simple Dimension, not across parting line.
2. Dimension across and perpendicular to parting line.
3. Dimension across but parallel to parting line (includes mismatch).

D2-2786-8

K1

SECTION K

APPENDIX

STATISTICAL ANALYSIS TECHNIQUES

Conventional statistical analysis methods were used to establish composition limits, design allowables, and dimensional tolerances. These methods are based on the assumption that the various data being studied represent a "normal" distribution. The following symbols are used:

- n = the number of individual values in a sample
- X = an individual value in a sample
- $\sum X$ = the sum of the individual values in the sample
- \bar{X} = the arithmetic mean (average) of the individual values in the sample
- σ = the standard deviation of the sample

The procedure for determining standard deviation is as follows:

1. Determine n and \bar{X} .

$$\bar{X} = \frac{\sum X}{n}$$

2. Determine the difference of each value from the average $(\bar{X} - X)$ -- n values will be obtained. Signs can be disregarded.
3. Square each of the values obtained in step 2 -- $(\bar{X} - X)^2$ -- n values will be obtained.
4. Determine the sum of the values obtained in step three and divide this sum by n .

$$\frac{\sum (\bar{X} - X)^2}{n}$$

5. Determine the square root of the value obtained in step 4. This will be one standard deviation.

$$\sigma = \sqrt{\frac{\sum (\bar{X} - X)^2}{n}}$$

BIBLIOGRAPHY

The following references were selected as pertinent to the objectives and investigations covered by this developmental program. Many of the listed references include further bibliographies, which will provide detailed background information for those interested in related research and development work.

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