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## EFFECTS OF TEMPERATURE-TIME-STRESS HISTORIES ON THE MECHANICAL PROPERTIES OF AIRCRAFT STRUCTURAL METALLIC MATERIALS

Part II. STRESSED EXPOSURE OF 7075-T6

C. D. Brownfield D. M. Badger

Northrop Corporation

SEPTEMBER 1960



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WRIGHT AIR DEVELOPMENT DIVISION

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WADC TECHNICAL REPORT 56-585 PART II

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C D. Brownfield D. M. Badger

Northrop Corporation

SEPTEMBER 1960

Materials Central Contract No. AF 33(616)-5769 Project No. 7360

WRIGHT AIR DEVELOPMENT DIVISION AIR RESEARCH ANI) DEVELOPMENT COMMAND UNITED STATES AIR FORCE WRIGHT-PATTERSON AIR FORCE BASE, OHIO

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#### FOREWORD

This report was prepared by the Norair Division of Northrop Corporation under USAF Contract No. AF 33(616)-5769. This contract is a continuation of a previous contract, AF 33(616)-3028. This contract was initiated under Project No. 7360, "haterials Analysis and Evaluation Techniques", Task No. 73404, "Fatigue and Creep of Materials." The program was administered under the direction of the Materials Central, Directorate of Advanced Systems Technology, Wright Air Development Division, with Mr. k. F. Klinger acting as project engineer.

This report covers work conducted from May 1958 to September 1959.

Norair personnel responsible for this program included Messre. D. M. Badger, C. D. Brownfield, and J. V. Griffin. The contributions of Metal Control Laboratories of Huntington Park, California, in their capacity as testing subcontractor are gratefully acknowledged. Primary technical personnel on this program at Metal Control Laboratories consisted of Messre. R. E. Clark and J. W. Vinatieri.

Acknowledgement is also given to the authors of Part I for their constructive suggestions in this study.

This report is identified at Morair as MOR 60-16.

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#### ABSTRACT

A study has been made on the problem of predicting strength of a har ened metal alloy after subjection to variable thermal and stress environments severe enough to cause permanent loss of properties. Methods have been developed for predicting tensile ultimate, tensile yield, and compressive yield strengths of 7075-T6 aluminum alloy after single or multiple exposures to various conditions of temperature and stress. An analytical expression suitable for automatic computing machine use has also been developed.

The results of tensile and compressive tests on alclad 7075-T6 aluminum alloy showed that stresses large enough to produce inelastic creep strain during thermal exposure cause reduction in residual strength after exposure. The test results have been used to establish the usefulness of the Larson-Miller exposure parameter for correlating residual strength after simple and complex, stressed and unstressed exposures.

#### PUBLICATION REVIEW

This report has been reviewed and is approved.

FOR THE COMMANDER:

W. J. TRAPP Chief, Strength and Dynamics Branch Metals and Ceremics Laboratory Materials Central

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		Units
D	$ \begin{array}{c c} \frac{r_{ef} - r_{1}}{r_{f} - r_{1}} & , \mbox{ strength deterioration factor describing } r_{ef} \\ r_{f} - r_{1} & in \mbox{ terms of its relationship to the strength } \\ \mbox{ interval between } r_{f} \mbox{ and } r_{1} \mbox{ at any temperature } T_{f} \mbox{.} \end{array} $	
Ff	Strength (Ftu, Fty, or $\mathbb{P}_{CY}$ ) at any temperature Tf with no previous exposure to temperature.	psi
l'ef	Strength ( $F_{tu}$ , $F_{ty}$ , or $F_{cy}$ ) at any temperature T <sub>f</sub> after exposure to T <sub>e</sub> .	p <b>si</b>
F <sub>1</sub>	Strength ( $F_{tu}$ , $F_{ty}$ , or $F_{cy}$ ) at the given temperature after exposure to a reference exposure condition.	psi
Ftu	Tensile ultimate strength	psi
F <sub>ty</sub>	Tensile yield strength	psi
F د <b>y</b>	Compressive yield strength	psi
RT	Strength reduction factor combining exposure tempera- ture effects and test temperature effects. Ratio of strength-at-temperature-after-exposure to original- room-temperature-strength.	
R <sub>€</sub>	Strength reduction factor for effect of stress during exposure. Ratio of strength-after-stressed-exposure to strength-after-unstressed-exposure at the same value of 9.	
R.T.	(Subscript) room temperature	
T <sub>e</sub>	Teaperature of exposure	°F or °R
Tſ	Temperature of strength test or of design condition after exposure.	<b>or</b>
t	Time	hours
•	Larson-Hiller exposure parameter expressing equivalent combinations of exposure temperature and exposure time. $\Theta = T (C + \log_{10} t)$	(T in <b>%</b> )
• <sub>17</sub>	$T(17 + \log_{10} t)$ , the specific variation of 9 used in this report.	(T in <sup>o</sup> N)
01	$T(20 + 1.46 \log_{10} t)$ , the variation used for 7075-T6 in Part I.	(T in <sup>o</sup> R)
£	Inelastic strain remaining after creep exposure.	\$

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INTRODUCTION

Modern flight vehicles are sometimes subjected to such severe thermal environments that the materials of which they are constructed suffer permanent loss in strength. Typically, these materials are hardened metal alloys which progressively lose strength during periods of exposure to temperatures at which the hardening mechanism is unstable. If stress is also present during the thermal exposure, as is often the case, this can have a further damaging effect on strength. Efficient design for such severe temperatures therefore requires a knowledge of how to predict the residual strength of structural materials after exposure to various histories of temperature and stress.

In the study reported here, empirical methods were established for predicting residual strength of aluminum alloy 7075-T6 in tension and in compression after such exposures. These methods are extensions of the approaches established by Fortney and Avery in Part I on aluminum alloys 7075-T6 and 2024-T3. The choice of 7075-T6 for the present study was made recognizing that this material is finding considerable use in moderate temperature applications and some use for limited times in higher temperature applications. Cost factors often make an aluminum alloy the best choice of materials for severe thermal and stress environments, providing accurate strength data are available. The methods of analysis developed in this study allow the use of the subject material to its practical limit in high temperature design.

In the most general form, these methods are based on curves which attempt to describe the characteristic deterioration or retrogression of strength with exposure, from fully hardened values toward annealed values. For unstressed exposures of 7075-T6 it was found that a single curve could be used to adequately describe characteristic deterioration of all strength properties considered (tensile yield and ultimate, and compressive yield) throughout most of the exposure range. For stressed exposures the same approach was used with modifications. Analytical methods for design use based on these curves are given in a special section presented early in this report.

For many other hardened materials it is considered likely that similar characteristic curves can be generated by applying the methods developed in this study. For materials which respond to thermal exposure in a more complex fashion than 7075-T6 or which have a markedly different response to stress during exposure, these methods would have to be modified, possibly toward a less general type of final curve.

The empirical approach used to define the characteristics of 7075-T6 in this study was at follows: .063 inch Alclad sheet coupons were subjected to a variety of exposure conditions and then tested at selected temperatures to determine residual short-time strength. Stressed exposures were studied by applying tensile stress during exposure with the amount of stress chosen to produce up to 1 percent inelastic strain --- about the practical limit for design based on short-time strength after exposure.

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The scone of the single exposure was:

Temp. range	250-600°F
Time range	1-100 hr
Stress range	0-50,000 psi (tension)*

The scope of multiple exposures was:

Temp, range	275-400°F	
Time range	3.5-340 hr	
Stress range	to 1 percent accumulated inelastic st	rain

The variety of multiple exposures included:

1-step, 2-step, 3-step, 4-ste, and 10-step sequences
Increasing temperature sequences
Decreasing temperature sequences
Mixed sequences
Stressed sequences to various strain levels
Stressed sequences with zero-stress soaking periods at start or end

The scope of final test conditions after exposure included:

Several mechanical properties:	F <sub>tu</sub> , F <sub>ty</sub> , F <sub>cy</sub> , percent elongation, hardness
Several test temperatures:	R.1., 200, 500, 400 r
One strain rate:	.01 in./in./min. to yield
then:	.0105 in./in./min.

The presentation of information in this report is oriented from the standpoint of design use. Major results are presented first followed by substantiation and detailed descriptions. Illustrative material is grouped together at the end of the report for convenience in following references frequently repeated throughout the report. This places the graphs of final results (Figure 17 through 20) which are most useful for design applications just inside the back cover where they can be found quickly.

\*Note: A spot-check of the effect of compressive exposure stress was also included.

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Empirical methods for predicting residual strength of two aircraft structural materials after exposures to temperature alone, without stress have been developed in the first phase of this work, reported by Fortney and Avery in WADC Technical Report 56-585, Part I. The subject materials were 2024-T3 and alriad 7075-T6 aluminum alloys, and strengths in tension only were considered. The use of the Larson-Miller time-temperautre parameter for predicting the total effect of a sequence of differing unstressed exposures was successfully demonstrated in the aforementioned report. Extension of these methods to the case of s'ressed exposures and to design mechanical properties other than yield and ultimate strengths in tension was the purpose of the study reported here. Other studies, on alloys other than 7075-T6, have considered effects of prior creep on subsequent strength for creep strains ranging above about 0.5 percent (References 4, 5, and 6). None of these referenced studies were primarily concerned with development of methods for accounting for prior creep exposure effects in practical design cases.

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A number of valuable conclusions can be drawn from the results obtained in the study. The limitations of these conclusions are cited in the text.

- 1. Stress applied during severe thermal exposure can significantly accelerate deterioration of strength in 7075-T6 (retrogression of strength level toward the level of the annerled state).
- 2. The degree of strength reduction due to exposure stress is associated with the amount of inelastic strain accumulated during exposure. It is dependent upon the total degree of thermal exposure but is affected only slightly by manner of exposure accumulation for the cases investigated.
- 3. The Larson-Miller exposure parameter can be used to define equivalent single exposures for determining residual tensile ultimate, tensile yield, and compressive yield strengths after single or multiple stressed or unstressed thermal exposures.
- 4. The general strength deterioration characteristics of 7075-T6 can be adequately represented on a single curve. Tensile ultimate, tensile yield, and compressive yield strengths and all final test temperatures are reflected in this curve. For stressed exposures, separate curves are required for different degrees of straining during exposure.

Based on these findings, analytical methods for determining allowable strengths in practical design situations have been developed. These are presented in the following section, together with limitations on their usage. Substantiation of both the conclusions and the methods can be found under experimental results.

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#### ANALYTICAL METHODS FOR DESIGN

In practical design situations it is sometimes necessary to predict allowable short-time strength after complex histories of temperatures severe enough to cause partial annealing and of stress severe enough to cause creep. "Short-time" or "static" failures as opposed to "creep" failures are a likelihood for environments in which there is considerable variation in the severity of stress-temperature cycles, and in which the severe stresses are of short duration. Such environments may or may not produce amounts of creep significant from the standpoint of design.

Analytical procedures for assessment of strength under such conditions are presented here. In the approach used, the failure is artifically separated from the spectrum, and thereby the effects of the prior history on the allowable strength remaining may be defined. The prior history is simplified by computing an equivalent exposure value for the temperature spectrum alone and by accounting for the effects of creep separately.

To find equivalent thermal exposures, the Larson-Miller exposure parameter,  $\Theta$ , is used:

$$\Theta = T(C + \log t) \tag{1}$$

where T is temperature in degrees Rankine, t is time in hours, and C is a constant. For the 7075-T6 data used in this report, the best value of C has been found to be 17, and the parameter becomes:

$$\Theta_{17} = T(17 + \log t)$$
(2)

Given an actual spectrum of temperature exposures, the first step in computing the equivalent total exposure for 7075-T6 is to reduce each increment of exposure in the spectrum to a value of  $\Theta_{17}$  using equation (2). Next, a single reference temperature,  $T_{ref}$  is chosen arbitrarily, preferably within the temperature range of the spectrum. The equivalent time at this temperature is computed using equation (2) for each of the previously determined values of  $\Theta_{17}$ . Adding the results gives the total equivalent time at  $T_{ref}$  for the entire spectrum, as shown below:

Given Spectrum	<u>0</u>	Find Time at Tref	
T1, t1	(9 <sub>17</sub> ) <sub>1</sub>	Equiv t <sub>l</sub>	
<sup>r</sup> 2, <sup>t</sup> 2	(9 <sub>17</sub> ) <sub>2</sub>	Equiv t <sub>2</sub>	
T3, t3	(• <sub>17</sub> ) <sub>3</sub>	Equiv t3	
•	•	•	
•	•	•	
•	•	•	

Total Equiv. Time

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The equivalent total thermal exposure,  $\theta_{17}$  is now computed directly from equation (2) using the total equivalent time and the reference temperature.

The evaluation of a spectrum can be considerably simplified by using Figure 21. With this diagram the only calculation required for the above example is addition of the equivalent times at  $T_{ref}$ .

Once the equivalent total exposure is known, the assessment of its effect (if any) on strength can be made. It is useful to separately identify the major sources of strength reduction encountered. Strength is reduced as a result of: (1) duration of exposure to a sufficiently high temperature and (2) temperature at the time of failure. Combining these two effects, the residual strength after temperature exposure and at a given test temperature can be expressed as a function of original room temperature strength, as follows:

$$F_{ef} = R_T F_{R_*T_*}$$
(3)

(~)

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where  $F_{ef}$  is residual strength ( $F_{tu}$ ,  $F_{ty}$ , etc.),  $R_T$  is a strength reduction factor combining exposure temperature effects and test temperature effects and  $F_{R-T}$  is the original room temperature strength.

Stress during exposure, especially if high enough to cause inelastic strain, is another factor which may affect residual or final strength. It will be seen that stress did have a significant effect on the 7075-T6 material tested in this work. Including this factor, equation (3) above becomes,

$$F_{ef} = R_T R_{\varepsilon} F_{R,T}.$$
 (4)

where  $R_{\varepsilon}$  is a strength reduction factor for the effect of strain accumulated during stressed exposures.

Values of  $R_T$ ,  $R_{\xi}$ , and  $R_T R_{\xi}$  obtained for 7075-T6 are shown in Figures 16 and 17 plotted against exposure value  $\Theta_{17}$ . These were obtained from a program of single exposure tests and checked at key points by program of sequential exposure tests.  $F_{tu}$ ,  $F_{ty}$  or  $F_{cy}$  values may be determined for a variety of exposure conditions and test temperatures using these data and equation (4). The  $F_{R.T.}$  value used is the corresponding allowable room temperature  $F_{tu}$ ,  $F_{ty}$  or  $F_{cy}$  (see Reference 7). Limitations on the use of these data are discussed at the end of this section.

It can be readily seen that the curves of Figure 17 are not sufficiently general in application to cover all test temperatures and exposure conditions which are practical for this material. A more general strength deterioration factor has been developed for 7075-T6 which relates all three types of strength  $F_{\rm tu}$ , Fty, and  $F_{\rm cy}$  (and possibly others not tested) at all test temperatures. This factor is shown plotted in Figure 20(a) against exposure value  $\Theta_{17}$ . In general terms, it represents a relative strength value where 1.0 is the fully hardened strength and zero is a reference (nearly annealed) strength. This strength deterioration factor, D, is:

$$D = \frac{F_{ef} - F_1}{F_f - F_1} | T_f$$
(5)

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in which  $F_{ef}$  is the strength as tested after a given exposure,  $F_{f}$  is the original (fully hardened) strength,  $F_{1}$  is the reference (nearly annealed) strength, and  $T_{f}$  is test temperature. For a given value of D, the  $F_{ef}$ ,  $F_{f}$  and  $F_{1}$  values must all be the same type of mechanical strength ( $F_{tu}$ ,  $F_{ty}$ , or  $F_{cy}$ ) and obtained at the same test temperature,  $T_{f}$ .

When Figure 20(a) is used as a design curve,  $F_{ef}$  is the unknown residual strength to be determined. Rearranging equation (5) for this purpose gives,

$$\mathbf{F}_{ef} = \mathbf{D} \mathbf{F}_{f} + (1-\mathbf{D}) \mathbf{F}_{1} | \mathbf{T}_{f}$$
 (6)

The appropriate value of D is obtained from Figure 20(a) for the particular total exposure value  $\theta_{17}$  (obtained as before). Appropriate values of  $F_f$  and  $F_1$  are chosen from Figure 19.

Notice that  $F_{ef}$  here is for the unstressed exposure case if D is taken from Figure 20(a). The same equation applies to stressed exposure cases if appropriate values of D are substituted. These are provided in Figure 20(b) for two stressed exposure cases involving, respectively, 0.2 percent and 1.0 percent inelastic strain accumulated during exposure. The proper value of D from Figure 20(b) is used in equation (6) above with appropriate values of  $F_f$  and  $F_1$  from Figure 19 (as in the unstressed case) to find  $F_{ef}$  for a stressed exposure case.

No further explicit accounting for the effects of stress during exposure is necessary. For illustration, assume a design which meets the usual aircraft design criteria of 0.2 percent inelastic strain, as substantiated by methods not covered in this report. Then, any effects on residual strength of a spectrum of conditions within the design envelope are covered by use of the D value for 0.2 percent inelastic strain. Limitations are discussed at the end of this section.

For automatic computing purposes, an analytical expression is usually preferable to an empirical curve. Referring to Figure 20(a), an expression which fits the 7075-T6 data for unstressed exposure quite well throughout most of the exposure range and which relates D to  $\Theta_{17}$  is:

$$D = \frac{34,800}{(\theta_{17} - 12,920)^4 + 28,950} - 0.202$$
(7)  
(13,000  $\leq \theta_{17} \leq 17,500$ 

This value of D can be used in place of Figure 20(a) for unstressed exposures and  $F_{ef}$  can be found as before.

The added influence of stress during exposure can be taken into account by a simple correction which will suffice for many practical design situations. For example, assume that the design criteria restricts accumulated inelastic strain to 0.2 percent. From Figure 20(b) it can be estimated that for a considerable range of exposure values above  $\Theta_{17} = 14,500$ , the effect of this

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d gree of inelastic strain is the same as adding a small increment of exposure value, approximately

 $\Delta \theta_{17} = 100$ 

The new value of D can then be found using equation (7) above with the corrected value of  $\theta_{17}$ , in this case  $\theta_{17} + 100$ , as

$$D = \frac{34,800}{(\theta_{17} - 12,820)^4 + 28,950} - 0.202$$
(8)  
(14,500  $\leq \theta_{17} \leq 17,500$ )

If accumulated inelastic strains significantly larger than 0.2 percent are permitted, a larger correction must be made. To correct for 1.0 percent strain, the appropriate increment of  $\Theta_{17}$  can be estimated from the curves of Figure 20(b) for each mechanical property. In many calculations, an average correction for the three properties will suffice and will simplify the calculations. In this case, for  $\Theta_{17} \ge 14,500$ ,

 $\Delta \theta_{17} = 250$ 

may be used in equation (7) as in the previous case, and:

$$D = \frac{34,800}{(\theta_{17} - 12,670)^4 + 28,950} - 0.202$$
(9)  
(14,500  $\leq \theta_{17} \leq 17,500$ )

For  $\theta_{17}$  values between 13000 and 14,500 the appropriate curve should be chosen from Figure 20(b) to suit the design conditions. For automatic computing purposes the chosen curve can be represented either analytically as was done above or as a set of coordinate values.

The effect of inelastic strain beyond 1 percent is not considered since prior creep strains of this magnitude are not likely to be permitted (except in very local areas) in cases where short-time strength is required after creep exposure.

It should be noted that the inelastic strain values of 0.2 percent and 1.0 percent used in this report are the strain accumulated during exposure only. The yield strength tests produce an additional inelastic strain of 0.2 percent. Thus, yield strength curves for 0.2 and 1.0 percent strain during exposure actually represent 0.4 and 1.2 percent total inelastic strain, respectively. The exposure stress was considered separately to provide a consistent treatment of yield and ultimate strength results. For a vehicle designed to criteria restricting inelastic strain to 0.2 percent total from any and all sources, the above methods would be conservative.

The above methods have some limitations and also possibilities of extension. These are outlined below.

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Figure 20(b) shows that there is considerable difference between the stressed exposure curves for  $F_{ty}$ ,  $F_{tu}$ , and  $F_{cy}$ , for exposures below  $\theta_{17} = 14,000$ . Here there is a very noticeable Bauschinger-type effect, which has a greater influence on strength than the factor of acceleration of the aging process through straining. In this region, only two exposure conditions were studied and the effect is only approximately defined. However, this effect is usually not taken into consideration in design except when yield strength is reduced (i.e.: when direction of testing is opposite to that of the original straining). The compressive yield curve for 1.0 percent strain can be considered an approximate mate minimum curve for cases of this type, providing stresses during exposure are no more severe (in causing strain) than used in the  $300^{\circ}F-1$  hour condition tested.

Material variation must be considered in applying the results of this study. The curves of Figures 17 and 20 are representative only of the material tested in this investigation and should not therefore be used directly as minimum design curves. There is significant variation in the response of various lots of the subject alloy to exposure in addition to the normal variation in room temperature properties (see pg. 17). A suggested method of reducing strength values obtained from these curves to minimum allowable strengths is by reference to minimum design curves for 7075-T6 such as provided in Reference 7 (Mil Handbook-5). Comparison between equivalent data from this report and from the minimum design curves will indicate necessary adjustments in the curves of Figures 17 and 20. A better solution would be the development of true minimum curves of D versus Q. This could be done using the methods developed in this investigation, and, for the most part, existing data on 7075-T6.

It is felt that justification exists for cautious extension of most of the results to the case of compressive creep exposures. In the compressive creep exposure check test, 0.3 percent compressive creep strain had about the same effect on compressive yield strength as a like amount of tensile creep in the same intermediate exposure range (see Table VIII). It is not certain that like agreement would be obtained for larger amounts of strain. Also, it is probable but not certain that the Bauschinger-type effects noted for exposure values below  $\theta_{17} = 14,000$  would be of the same magnitude for straining in compression as in tension. If Figure 20(b) is used to estimate the effect of compressive creep during exposure, the compressive yield curve should be used as the tensile yield curve, and vice-versa. This is to account for the reversal of Bauschinger-type effects.

Investigation of shear and bearing strengths was beyond the scope of this present work. However independent studies (not reported) using 7075-T6 shear and bearing data from deference 8 have shown promise for application of Figure 20 to these failure modes. Until this is verified, shear and bearing strengths for unstreased exposures can be found from appropriate ratios to tensile values obtained for the given design conditions. It should be noted that these ratios are affected by test temperature and by approximate degree of exposure. Further studies of shear and bearing strengths after stressed and unstressed exposures are planned for further extension of this work.

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Care should be exercised in applying the results of the study to conditions outside the scope of the test program. High rates of straining followed by testing in the direction opposite to that of the straining is one possible area for caution. another such area is that of multiple tensile stressed exposures followed by testing in compression. In tests of this type, compressive yield strength was depressed slightly beyond expected values, most noticeably in the ten part sequence tests. It is not certain whether further strength loss might be caused by increasing numbers of cycles or even whether the effect is real. Care is therefore recommended in extending these results to much more complex sequences followed by compressive loading. For unstressed exposures, and for tensile properties following tensile stressed exposures the results indicate that extension to more complex sequences than those tested should be safe.

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The foregoing conclusions and methods are based on the results of the test program outlined in Tables I and II. The purposes of the various tests are discussed below, then detail; of the program and the results obtained are presented and reviewed under the following sub-headings, in conjunction with the tables and graphs at the end of this report:

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- . Test Materials and Basic Properties
- . Creep Characteristics of Material
- . Tabulated Results
- . Material Response to Thermal Expousre Without Stress
- . Exposure Parameter Re-evaluation
- . Effects of Stress During Single Exposures
- . Effects of Stress During Multiple Exposures
- . Variation of Stress Effects with Degree of Exposure
- . Residual Strength After Single Exposures
- . Accuracy of Prediction of Multiple Exposure Results
- . Generalization of Results

Each test consisted of two principal parts for all exposure conditions, single or sequential:

<u>Exposure of Septimens</u> -- Specimens were subjected to one of the selected temperature-time, or temperature-time-stress exposure conditions or schedules of conditions.

<u>Strength Testing After Eccosure</u> -- Standard short-time tensile and compressive tests were performed at room temperature or at one of the selected test temperatures.

The basic tests are the single exposure tests given in Table I. These tests provide data on stressed exposures, and they also form the foundation for examination of multiple exposures. The multiple exposure test program is presented in Table II.

Exposure temperatures, times, and stress levels shown in Table I were chosen to survey the effects of stressed exposure throughout the unstable range of the material from the T-6 condition to the nearly annealed condition. They were also chosen with the possibility in mind of using the Larson-Miller exposure parameter for stressed exposure cases as was done in Part I for unstressed exposure cases. In two cases, combinations of exposure time and temperature were approximately matched in total exposure value by two other combinations; this was done to check the assumed value of the constant C in the Larson-Miller exposure parameter. Values of this parameter using C = 17 (determination of which is discussed later) are presented for each suposure condition in the first column.

Exposure stress levels shown were established through creep tests and creep correlation methods. The target inelastic strain values were chosen to cover the range of anticipated practical use in design -- from a lower value

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at half of the commonly used 0.2 percent to an upper value of 1.0 percent. Tensile exposure stress was used throughout to study stressed exposure relationships, and a spot-check case of compressive exposure stress was included.

The final test temperatures after exposure were selected to cover the temperature range of practical usage of the alloy with a minimum of tests as assisted by generalizing procedures.

Table II presents the multiple exposure tests. These tests provide data on a wide variety of sequential exposure conditions. The descriptions given in the table identify the main characteristics of the sequences chosen. Increasing and decreasing temperature trends at low, high, and zero stress levels at key values of total exposure are represented, as are some special cases with zero stress in the first or last step. Sequences range from two steps to ten steps.

Exposure temperatures and times were chosen to provide the desired total exposure and also to cause a significant change in properties in each step. The upper and lower limits of time used were 100 hrs. and 3.5 hrs., respectively, and temperatures ranged from 275°F to 400°F. Sequence total exposure values were chosen to approximately match the single exposure cases which had been checked by two different combinations of temperature and time.

Sequential exposure stress levels were established to give the indicated target strain values in the total sequence and also to produce significant strain in each of the stressed exposure steps.

Test temperatures after sequential exposures duplicated those of the single exposure program and were chosen with similar objectives. However, it was found that reductions in the number of final test temperatures could be effected without compromising objectives.

#### Test Material and Basic Properties

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Test material for the entire program was 7075-T6 alclad aluminum ally. Three .063 inches thick, 48 x 144 inch sheets of this material in the as-supplied T6 temper were used from Northrop Corporation production warehouse stock. All were from the same box of sheets from the material supplier.

The chemical analysis of each of the sheets, shown below, was well within the applicable specification (QQ-A-287) limits. Analyses were performed on a direct reading spectrograph.

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#### PERCENT OF ALLOYING ELEMENTS

Element	QQ-A-287 Limits	Sheet A	Sheet B	Sheet C
Copper	1.2-2.0	1.70	1.60. 1.68	1.63
Magnesium	2.1-2.9	2.76	2.66. 2.58	2.70
Zinc	5.1-6.1	5.40	5.52. 5.48	5.50
Chromium	0.18-0.40	0.22	0.22, 0.21	0.22
Iron		0.10	0.18, 0.19	0.15
Silicon	0.50 max	0.06	0.05, 0.06	0.07
Manganese	0.30 max	0.10	0.09, 0.09	0.08
Titanium	0.20 max	0.02	0.02, 0.03	0.04
Aluminum	Balance	Balance	Balance	Balance

The microstructure of the test material was found to be typical of 7075 in the T-6 condition. Thickness of cladding was approximately 4 percent of the sheet thickness per side. A number of room temperature tensile tests were performed on as-received (unexposed) material from each sheet. All were well within the requirements of QQ-A-287.

Room and elevated temperature tensile and compressive properties of the test material before exposure are presented in Table III. The prefixes A, B, and C in the specimen nomenclature denote sheets A, B, and C which were utilized for the single exposure, tension; the multiple exposure, tension; and the compression program tests, respectively. The room temperature tensile tests performed on sheet A represent a survey of the sheet tensile properties. A total of 36 tensile tests were performed, to determine sheet variation and to provide tests to which nearby specimens exposed and tested could be compared. The latter improved accuracy in evaluating strength reduction factors in cases where strength values were near the original unexposed strength. The survey results are also shown in Figure 11, relating test values to location in the sheet. Variation in sheet A tensile properties was about average. The mean strength level was close to an universal mean for the same alloy and gage.

The average room temperature tensile properties of sheets B and C were approximately the same as sheet A. Room temperature compressive properties were determined only for sheet C. The average compressive yield of sheet C was 8.6 percent higher than its tensile yield, which is about normal.

Elevated temperature tensile and compressive tests in Table III are for unexposed specimens that have been heated to test temperature in a standard time of 12 minutes, from start of heating to start of loading. These data are shown plotted against temperature in Figure 19 ( $F_{\rm f}$  curves).

#### Creep Characteristics of Material

Some knowledge of the creep characteristics of the material used in this study was needed to establish stress levels which would give the target values of strain during stressed exposure conditions. This need was filled by data from the literature and by check tests on each sheet of material.

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Further refinements were required in methods for determining stress levels for succeeding steps in multiple exposure conditions. This was necessary because it was found that the creep strength of the material tended to decrease with progressive softening due to exposure. The strain values achieved during the various stressed single and multiple exposures are recorded in the tables of results. Analysis of creep characteristics is beyond the scope of this report.

#### Tabulated Results

The results of the experimental program are listed in Tables IV through XVI in the following order:

Table	Exposure Condition	Final	Nominal Test
No.	(Stressed & Unstressed)	Test	Temperature
IV	Single Exposure	Tension	R.T.
V	Single Exposure	Tension	200°F
VI	Single Exposure	Tension	300°F
VII	Single Exposure	Tension	400°F
VIII	Single Exposure	Compression	R.T.
IX	Single Exposure	Compression	200°F
X	Single Exposure	Compression	300°F
XI	Single Exposure	Compression	400°F
XII	Sequential Exposures	Tension	R.T.
XIII	Sequential Exposures	Tension	300°F
XIV	Sequential Exposures	Tension	400°F
xv	Sequential Exposures	Compression	R.T.
XVI	Sequential Exposures	Compression	300°F

The single exposure tables describe for each specimen the nominal and actual exposure conditions, the actual temperatures of testing after exposure, and the test results obtained. Actual exposure conditions consist of actual exposure temperature (the average temperature of the entire exposure) and the actual total inelastic strain obtained. The values of Larson-Miller exposure parameter  $\theta_{17}$  are also shown.

Similarly, actual test temperatures are distinguished from nominal; it was found that at 400°F very small deviations in test temperature could be significant, also there were a few cases where "room temperature" was significantly above the 68°F to 75°F range.

In addition to the strength values actually obtained in tensile and compressive tests, values corrected to the nominal exposure and test temperatures are provided. This was done as a guide in evaluating the effect of temperature deviations and to give a better basis for comparing results. Correction consisted of adjusting test values by an increment of strength derived from final curves of the test results themselves and appropriate to the error in exposure value ( $\theta_{17}$ ) or test temperature. This is felt to provide a reasonably accurate correction because of the large amount of data available.

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The strength results are also expressed in nondimensional form in the far right hand columns of these tables. The first of these,  $R_T$  is the ratio of the test value obtained to the material original unexposed room temperature strength. The second,  $R_{\varepsilon}$ , is the ratio of stressed exposure test results to the parallel unstressed exposure results, or a reduction factor for the effect of stress alone in the exposure environment.

The sequential exposure tables follow the same general pattern as the single exposure tables, but modifications are made to accommodate the multiple exposure steps. Nominal exposure conditions are shown; actual exposure temperatures are omitted but were used to compute the total exposure value  $\theta_{17}$  listed for each specimen. Actual strains accumulated during each exposure step are shown below the nominal conditions. Actual strains accumulated during the entire sequences are also presented. The properties after exposures and strength ratios are treated in the same manner as in the single exposure tables.

#### Material Response to Thermal Exposure Without Stress

The results of the single exposure tension tests are plotted in Figure 12 together with similar results from Part I to compare the response of the two lots of material to thermal exposure. The form of the Larson-Miller parameter,  $\Theta^*$ , established for 7075-T6 in Part I is used as the abscissa. Values of this form of the parameter are not shown in the tables because another form of the parameter is used for the balance of this report. However, values of  $\Theta^*$  can be readily computed from the equation given in Figure 12. The ordinate is non-dimensional strength as established in Part I.

The test data plotted are from Sheet A of the present study and represent the averages of all identical tests. These data are connected with solid line curves, and the data from Part I are shown by dotted line curves.

Comparing the results from the two lots of material, it can be seen that the material from Part I responded to exposure significantly earlier in time (or in  $\Theta^*$ ) than did the material in Sheet A. Similar disagreement in proportional effect of exposure has been found in comparison of results from comparable exposure and test conditions from various sources (i.e.: Reference 8). This is a factor of material variability that must be considered in applying results of a study of this type of practical situations. Methods of approximately accounting for this variability are discussed under Analytical Methods for Design.

Another difference in results from the two studies is also illustrated in Figure 12. It was originally assumed that the  $\Theta^*$  form of the Larson-Miller parameter used in Part I would also apply to the material used in the present program. This assumption was checked by the tests in which different combinations of exposure time and temperature gave the same value of  $\Theta^*$ . Test data plotted in Figure 12 at these values of  $\Theta^*$  ( $\Theta^* = 17,400$  and 18,500) show that imperfect agreement was obtained with the  $\Theta^*$  form of the parameter. Considerably better agreement was obtained when the parameter was modified. This is discussed in the following sub-section.

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Exposure Parameter Reevaluation

Reexamination of the Larson-Miller exposure parameter showed that

$$\Theta_{1'7} = T(17 + \log t)$$

provided the best fit for the data from the material in the present study (the change to a single constant is discussed below). The results using the new constant in the parameter are shown in Figure 13. The better agreement for the ultimate strength of Sheet A is immediately apparent (the  $376^{\circ}F-3$  hr. data point should be ignored for the moment, since it is from Sheet C and was not shown in the preceeding figure). The yield strength curves show similar good agreement for both Sheet A (tension yield points) and Sheet C (compression yield points).

There is thus the indication that the best constant for use in the parameter may change with different lots of material. The degree of change in this case can best be seen by converting the earlier  $\Theta^{\dagger}$  into the single-constant form by dividing by 1.46:

$$\frac{\Theta^{*}}{1.46} = T(13.7 + \log t)$$

Thus, the change in constant is effectively from 13.7 to 17. The change to the single constant form of the Larson-Miller parameter in this report is made because this has become the more widely used form.

#### Effects of Stress During Single Exposures

The effects of stress applied during exposure can be most simply discerned by comparing results for stressed and unstressed exposures having otherwise identical conditions. This may be expressed in ratio form as:

$$R_{\varepsilon} = \frac{r_{stressed exposure}}{F_{unstressed exposure}}$$

 $R_{\varepsilon}$  is thus a strength reduction factor for the effects of stressed exposure, and the strain subscript  $\varepsilon$  is used because the value of this factor appears to be primarily dependent upon the amount of inelastic strain accumulated during exposure.  $R_{\varepsilon}$  is easily found from the test data because unstressed exposure control specimens were included with each group of three specimens subjected to stressed exposures. The control specimen test results are given in Tables IV through XI just above the associated stressed exposure results.  $R_{\varepsilon}$  is shown in the tables and is obtained by dividing the individual stressed exposure final strength values by the average of the control specimen final strength values.

Study of the ratios in the tables shows that exposure stress (or strain) in general does have an effect on remaining strength of 7075-T6, but this effect is not large.  $R_{\rm C}$  ranges from about .85 to 1.05 depending on degree of thermal exposure and degree of strain produced during exposure.

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Figure 14 shows the  $K_{\epsilon}$  values from the single exposure conditions plotted against inelastic strain. Each individual test result is plotted. Separate plots are made for each strength property measured and for each degree of exposure utilized. Results for all final test temperatures are plotted together. An interesting and useful observation can be made: Final test temperature after exposure has no significant effect on the reduction factor due to stress applied during exposure. This can be confirmed by careful examination of all cases plotted in Figure 14.

In Figure 14, rough scatter bands are indicated by shading to show the basic trends of the results. These scatter bands are drawn arbitrarily in view of the small net effect that precision of establishing these bands has on the final resulting strength. The shading of the bands is stopped just above 1 percent strain because that is the upper limit of the range of interest in this study. However, points with larger strain were considered in drawing the bands.

Included in these bands are key markings (small, black "x" marks) at 0.2 percent strain and at 1.0 percent strain for use in later analysis. These marks are plotted at the approximate mean value of the test data, to the nearest 0.01 value of  $R_{\epsilon}$ . A small degree of smoothing of the location of these marks across neighboring exposuring conditions was done (up to 0.01  $R_{\epsilon}$ ). The scatter of individual test points about the key points is within 3-1/2 percent for most cases.

At the lowest value of exposure, Figure 14(a), no scatter band is established because only large-strain tests were run. Comparing the key points shows that stress during exposure had no appreciable effect on the tensile ultimate strength but it raised the tensile yield strength and depressed the compressive yield strength. This is similar to the Bauschinger effect observed in specimens which have been prestrained in tension (at room temperature) and then tested either in tension or compression (Reference 10). The comparison seems logical since the exposure stress in the 300°F-1 hour case was quite high, and a relatively large part of the resulting strain was probably plastic strain on loading.

At moderate exposure values, Figures 14(c) through 14(g), this pronounced effect has disappeared and the scatter bands for all three properties are depressed, with compressive yield reduced slightly more than the tensile properties. In this area, the predominant effect of strain during exposure seems to be an acceleration of the process of everaging or annealing. However, a small degree of Bauschinger-type effect appears to persist to fairly high exposure values.

At extreme exposure values, Figure 14(h) and 14(i), the effect of stress during exposure on remaining strength is diminished and the  $R \in$  values again approach 1.0. In this exposure range, there may be a tendency for ultimate strength to be least affected and compressive yield the most affected.

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#### Effects of Stress During Multiple Exposures

The effect of creep stress during the sequential exposures is presented in Figure 15. The manner of presentation is similar to that of the single exposure results discussed in the previous section, so that visual comparisons may be made at nearly equivalent exposure values ( $\theta_{17}$  values). Direct comparisons of single and multiple exposure results are presented later in the report. In comparing the single and multiple exposure curves it is valuable to note that the vertical scale of these plots is a rather expanded one, so that any real differences should be easily detected.

Examination of the curves in Figure 15 shows that of the sequence variables considered. There is little evidence of greater strength reductions than found in comparable single exposure tests. The one sequence in which a noticably different result was obtained is the D sequence shown in Figure 15(e). In the D exposure sequence, specimens were unstressed until the last fraction of exposure period, at which time an exposure stress which gave a relatively high rate of straining was applied. Reductions in final strength caused by exposure strain in D sequence tests were not as great as in tests at similar values of strain and exposure value  $\Theta_{17}$ , in which the strain was more uniformly accumulated. The results from these tests indicate that the stressed exposure strength reductions can be affected to some degree by extreme non-uniformity in strain accumulation.

Slightly greater strength reduction than expected was obtained in compression tests after the stressed E sequence (10 Part) expouse, shown in Figure 15(f). It is not certain whether this effect is real since it is small in magnitude especially considering the number of test results defining it and their scatter. The results obtained raise questions as to the effect of further increases in number of steps in sequences. Additional tests would be required to definitely establish the nature of this effect.

The various types of sequential exposures serve to demonstrate that, except for unusual exposure conditions such as the D sequence, (and possibly the case of compression after very complex stressed exposures, as noted above) the results are independent of the type of sequence used. Thus, it may be assumed for a large number of practical problems that if the sequence is sufficiently well sixed to avoid radical situations, consistent results can be expected from all sequences in the range covered by this study.

#### Variation of Stress With Degree of Exposure

As noted above, the effect of stress during exposure on the subsequent residual strength varies with degree of exposure. Figure 16 shows this variation; the key points for 0.2 percent and 1.0 percent inelastic tensile strain from Figures 14 and 15 are replotted as a function of  $\Theta_{17}$  to show the trends of the mean values of the scatter bands with exposure.

The curves were drawn through the single exposure key points only (x symbol) for several reasons: (1) to establish a set of single exposure  $R_E$  values to be used in later analysis, (2) to provide a basis to which stress

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effects in multiple exposure tests could be compared, and (3) to provide a basis for some smoothing of single exposure key values across adjacent exposure conditions. Key points are omitted and dashed lines are used where  $R_{\rm C}$  curves had to be interplated or extrapolated to obtain values at either 0.2 percent or 1.0 percent strain. Multiple exposure key points (+ symbol) are plotted for comparison with single exposure results.

The maximum effect of exposure stress and the trend of this effect with single exposures are easily discerned from this figure, since the effects of scatter are minimized. It can be seen that for 1 percent accumulated tensile strain during exposure the maximum effect on residual strength is about 8 percent for tensile ultimate, 10 percent for tensile yield and 11 percent for compressive yield. At very early exposures, greater differences between the effects in the three cases are noted, as discussed earlier. For 0.2 percent inelastic strain, the maximum effects are about 3 percent of tensile ultimate, 4 percent of tensile yield, and 2 percent of compressive yield. The differences at 0.2 percent strain are considered smaller than the accuracy of definition of the trends.

It is of interest to note the shapes of the yield curves for 1 percent strain during exposure as compared to that of the ultimate tensile curve. The compressive yield minimum is both lower than that of the tensile yield and occurs earlier. While the tensile yield minimum even occurs later than that of the tensile ultimate. Apparently a small degree of Bauschinger-type effect persists to relatively high exposure values. Possibly, much shorter exposure periods than those investigated, with rapid accumulation of strain during exposure would emphasize this effect. However, it is significant that there was no apparent difference in this effect between 10 hour and 100 hour tests at approximately the same exposure value.

In Figure 16, the multiple exposure results follow much the same pattern as the single exposure results for tensile ultimate and yield strengths. Compressive yield, on the other hand shows a possible tendency toward more strength loss in multiple exposures than in single exposures, most noticeable in the 10 step sequence, as noted in the previous discussion. Comparison of single and multiple scatter bands at equivalent exposures for the compressive yield cases indicates that the difference is slight for the variety of cases covered. The cause of this difference, if a real effect, is not identifiable from these tests. Additional studies are desirable to check the existence of this effect.

The small net error introduced in predicting multiple exposure results for sequences within the scope of this investigation is treated in later discussions of the accuracy of prediction. The D sequence is the one case in which marked difference was obtained. In Figure 16, the D sequence points for tension ultimate and tension yield after 1 percent strain fall quite high, near the 0.2 percent strain curve, at  $9_{17} = 15,300$ .

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#### Residual Strength After Single Exposures

In the foregoing discussion, the strength reduction factor for stressed exposure has been based on the strength after unstressed exposures. Strength after unstressed exposure is reduced as a result of: (1) duration of exposure to a sufficiently high temperature and (2) temperature at the time of failure. These effects can be combined and the total strength reduction for unstressed exposure expressed as the ratio:

# $R_{T} = \frac{\text{Strength at Temperature After Unstressed Exposure}}{R_{*}T_{*}}$

For stressed exposure the reduction factor for the effect of stress,  $R_{\xi}$  is included in the expression for total strength reduction, as:

# $R_{T}R_{\epsilon} = \frac{\text{Strength at Temperature After Stressed Exposure}}{R.T. Strength Before Exposure}$

Expressed in these terms, stressed and unstressed exposure results can be plotted on the same coordinates, providing curves that are more general than obtained by plotting actual strengths.

Figure 17 shows the  $F_{tu}$ ,  $F_{ty}$  and  $F_{cy}$  strength reduction factors  $R_T$  and  $R_TR_{\epsilon}$  plotted as a function of exposure value  $\theta_{17}$ . Trends of  $R_T$  values are indicated by the solid curves.  $R_TR_{\epsilon}$  values for 1.0 percent strain during exposure are shown connected by dashed curves. Each  $R_T$  data point represents an individual unstressed exposure test result. These values are also listed in Tables IV through XI. Values of  $R_TR_{\epsilon}$  were obtained by applying  $R_{\epsilon}$  values for 1.0 percent strain from Figure 16 to the  $R_T$  curves.

Correlation of unstressed exposure results with the Larson-Miller parameter as modified for this study appears good, as was noted previously in discussion of Figure 13. The stressed exposure results indicate that although inelastic strain during exposure does affect strength, it does not impair this correlation. This can be seen by reference to Figures 14 and 16. As noted in previous discussion of these figures, in cases where different combinations of time and temperature gave approximately the same exposure value  $\theta_{17}$ , the effect of inelastic strain on final strength was nearly the same.

The general effects of inelastic strain during exposure can be seen by examination of the curves of Figure 17 for all strength properties and final test temperatures. First, it is observed that the effect of up to 1.0 percent inelastic strain on the residual strength of 7075-T6 is not large but is significant. Second, the effect is nearly the same in all cases at a given exposure value except in the lower exposure values. A third observation is that for a large part of the exposure range the stressed exposure results are offset from the unstressed exposure results by approximately a constant increment of exposure. These observations make possible a much simplified, general approach for predicting the effect of stressed exposures presented in this report.

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#### Accuracy of Prediction of Multiple Exposure Results

Use of the single exposure results to prodict multiple exposure values for the same total exposure has already been suggested for unstressed exposures in Part I and for stressed exposures in foregoing discussions. Comparison of the accuracy of such predictions is given in Figure 18 for the three strength properties investigated and for unstressed and stressed exposures. The actual results achieved in test after multiple exposures are plotted against values predicted from the single exposure curves for the same calculated total  $\theta_{17}$ exposure. The solid line at 45 degrees in each chart represents perfect agreement.

Unstressed exposure correlations are shown in Figure 18(a). Individual points reflect averages of all results for each sequence. Coding is provided to identify the different sequences investigated. Agreement between predicted and actual results is in general quite good. One difference can be detected between results from certain exposure sequences. Considering sequences identical except for the temperature direction of the exposure steps, decreasing temperature steps resulted in slightly higher strengths than increasing temperature steps. This can be seen by comparing the open and closed symbols having the same shape. Even with this difference, the agreement between predicted and actual results is good.

The stressed exposure correlations are shown in Figures18(b) and 18(c) for 0.2 percent and 1.0 percent inelastic strain. In this case the test values for comparison with predicted values were derived by using the key values of  $R_{\epsilon}$  (for 0.2 and 1.0 percent strain) for the stressed multiple exposure cases from Figure 15. Each key  $R_{\epsilon}$  value was multiplied by the  $R_{T}$  value for the sequence it represented. This removed the scatter of individual test results as was done for Figure 18(a). Also, the  $R_{T}$  values for sequences identical excepting for direction of temperature steps were averaged, removing the difference noted previously for different temperature directions of unstressed exposure sequences. (No distinction in stress effects was noted for this case, see Figures 15(a) and (c)). Since accuracy of the  $R_{T}$  value prediction in Figure 18(a) was good (especially with temperature direction effoct averaged), Figures 18(b) and 18(c) clearly indicate ary additional inaccuracy due to the effect of stress (or strain) during the various sequences.

The results for the stressed exposures are similar to those found for the unstressed exposures. The correlation between predicted and actual test results is generally very good. The maximum difference is noticed in the case of the D sequence. Another smaller difference can be detected. This is for compressive yield strength after the ten-part E sequence. Both of these results were discussed under "Effects of Stress During Multiple Exposures."

#### Generalization of Results

It is valuable to generalize the foregoing results in order to simplify and extend application to practical problems. This can be accomplished by finding ways to relate as many variables as possible to each other on a common basis. This has already been partly accomplished by the general correlation of all test results with the form of the Larson-Miller parameter used.

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Closer examination of the various curves in Figure 17 shows that all material strengths do not deteriorate to the same degree with prolonged exposure. Room temperature tensile ultimate strength declines to 50 percent of its original strength while room temperature tensile yield and compressive yield strengths decline to about 20 percent. This means that any relationship which exists between the magnitudes of these various strengths must be nonlinear and is not entirely obvious from the empirical data. However, these curves have some similarities. It can be seen that in all cases deterioration of properties begins to occur at a common degree of exposure and becomes essentially complete at another common degree of exposure. Similarity can also be seen in the general shapes of the curves. This leads to the supposition that a single general shape of decay curve might be followed by all strength properties in traversing the interval between full-hard strength and full-soft strength.

Attempts were therefore made to plot all properties tested in terms of relative strength between full-hard and full-soft strength, against the exposure parameter  $\Theta_{17}$ . This was not too successful, due to a rather large degree of scatter which occurred in tests at extreme exposures (see plotted data for 600°F-8 hours in Figure 17(c)). Attempts to approximate the same results by using a more restricted interval have been met with success. The interval between the original strength and the strength remaining after an exposure of  $\Theta_{17} = 16,380$  (450°F-10 hr.) was chosen as the reference interval. The strength values for the extremes of this interval (for the three sheets of material used in this study) are shown in Figure 19 as a function of temperature. All other strength values are related to their position in this interval by the expression:

 $D = \frac{F_{ef} - F_1}{F_f - f_1} |_{T_e}$ 

in which D is called the strength deterioration factor,  $F_{ef}$  is the strength as tested after a given exposure,  $F_{f}$  is the original (full hardened) strength,  $F_{1}$  is the strength after the reference exposure (450°F-10 hr.) and  $T_{f}$  is test temperature. For a given value of D, the  $F_{ef}$ ,  $F_{f}$ , and  $F_{1}$  values are all the same type of mechanical strength ( $F_{tu}$ ,  $F_{ty}$ , or  $F_{cy}$ ) and obtained at the same test temperature,  $T_{f}$ .

Values of D for unstressed exposures are computed in Table XIX using data from Tables IV through XVI for  $F_{ef}$  values, and Figure 19 for values of  $F_1$  and  $F_f$ . Fef values used are average values from each group of specimens tested under identical conditions. The results are plotted in Figure 20(a), where all of the unstressed exposure data from the program are shown together on one curve; all three types of mechanical properties at all final test temperatures after single and multiple exposures are represented. It can be seen that the correlation throughout most of the exposure range is good while that at extreme exposure (nearly fully annealed) is poor. Fortunately, the useful portion of the curve is the region of good correlation.

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The same procedure is applied to the stressed exposure cases by relating them to the unstressed exposure results. The  $F_f$  and  $F_1$  values are unstressed exposure values, while the  $F_{ef}$  values are for stressed exposures. Calculations for D are given in Table XIX for 0.2 percent accumulated inelastic strain and for 1.0 percent accumulated inelastic strain during exposure. In these pages,  $F_{ef}$  is found by applying the strength reduction factors,  $R_e$  from Figure 16 to the same unstressed exposure average values of  $F_{ef}$  used for the basic D curve.

The results for the stressed exposure cases are shown in Figure 20(b). Here, the data points are omitted for clarity of presentation. The curves for the different properties are practically the same for most of the unstable (steeply sloping) region of the curve and are shown as a single curve for 0.2 percent and another for 1.0 percent. At the low exposure values, the stressed exposure curves for the three properties separate, particularly the 1 percent strain curves. The magnitude of the effect is considerable below  $\Theta_{17} = 13,500$ . In this region the curves are primarily based on only one exposure condition (300°F-1 hr.) and it is not certain that the same curves would hold for all exposures.

All of the test data developed in this investigation have been successfully normalized by means of the procedures which have been discussed. This raises the question of extension to mechanical properties other than those tested. It should be recognized that the normalizing procedures used in this report are basically empirical, and therefore, the resulting strength deterioration curves apply specifically to those mechanical properties for which data were obtained. However, an analysis of data from reference 8 ( the results of which are not shown here) has indicated that the shear and bearing properties of 7075-T6 after various exposure conditions can very likely also be normalized by the same procedures. For unstressed exposures, shear and bearing properties may even follow the same strength deterioration curve (Figure 20(a)). The hardness data obtained in this investigation on the other hand, does not plot directly on the same curve, probably due at least in part to characteristics of hardness magnitude scales.

It is believed that the basic approach used in this work on 7075-T6 can also be used in investigating many other hardening materials which deteriorate in strength during exposure to practical thermal and stress environments. Modifications to the approach will no doubt be required, especially for materials having more complex aging or annealing characteristics than 7075-T6.

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The procedures used in this program were established and checked prior to use in an attempt to attain the accuracy desired in each part of the exposure and of subsequent tests. The consistency of the results achieved tends to verify the adequacy of the procedures. The important features of these procedures are discussed under the following subheadings:

- · Specimen Identification and Preparation
- . Specimen Exposure Assemblies
- . Specimen Exposures
- . Measurement of Creep Strain
- · Specimen Processing Between Exposure and Strength Testing
- . Hardness Testing
- Strength Testing

#### Specimen Identification and Preparation

Three 0.063 gage Alclad 7075-T6 aluminum alloy sheets were used during this investigation. Specimen blanks were marked out on heavy adhesive-backed paper covering the sheet, with the axis of all test specimens transverse to the original rolling direction. These were identified by metal stamping according to the identification system shown in Figure 1. The sheet was sheared into specimen blanks and machined to the dimensions shown in Figure 2. Careful machining practices were used, and microhardness tests on machined surfaces showed that machining caused no overheating of the material.

Specimer details are shown in Figure 2. The tension specimen incorporates pin-joint type loading ends, 0.505 inch reduced test section and 2.0 inch gage length. This same specimen was used for both creep-exposure and tensile testing after exposure. Tensile creep-exposure specimens for the compression test program were essentially extra long tensile specimens with a slight machining allowance on the width so that all edges of the compression specimens could be machined to the required tolerances. The compression test specimen was 0.5 in. wide by 2.75 in. long. Three of these specimens were machined from the reduced section of each creep-exposure specimen.

#### Specimen Exposure Assemblies

Stressed exposure of tension specimens was accomplished for the most part with three specimens linked in series as shown in Figure 3. Unstressed control specimens were exposed along with the stressed specimens, mounted (by one end only) adjacent to the center stressed specimen. Small aluminum blocks were loosely assembled to cover the reduced section of the upper and lower stressed specimens. The use of these blocks improved temperature stability with no detrimental effects other than a slightly slower heat-up rate. Thermocouples were placed at each end of the reduced section of each specimen until enough temperature data were accumulated to allow reduction of the number of thermocouples to that shown in Figure 3.

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Some of the one hour and ten hour stressed exposures of tensile specimens were effected in shorter length creep furnaces one specimen at a time. Thermocouples were located at each end of the reduced section of all such specimens. No temperature stabilization blocks were required.

All thermocouples were attached so as to contact the edge of specimens. This location was checked against thermocouples located in drilled holes in a dummy specimen and found to be an accurate procedure. (Temperature deviation between edge location and hole location was found to be less than 1°F as long as reasonably uniform furnace temperatures existed. For rapid heat-up conditions, the edge thermocouples responded more rapidly to increase in furnace air temperatures than did the thermocouples in holes.)

Stressed exposure of material for compression specimens was accomplished with the creep exposure specimen shown in Figure 2. Unstressed exposure specimens consisted in most cases of 0.625 in. by 3 in. blanks. These were mounted in the center of a second creep exposure specimen which acted as a carrier only. The carrier specimen was mounted (by one end) parallel to the stressed exposure specimen separated by about 0.25 in. In a few cases a long creepexposure specimen was utilized for unstressed exposure compression specimens, mounted the same as the "carrier" specimen described below.

A total of four thermocouples were used to instrument the ten-inch long reduced section of the stressed exposure specimen. A single thermocouple sufficed for the zero stressed specimens, all of which were in mutual contact.

#### Specimen Exposures

Most specimen exposures were performed in the type of creep machines shown in Figure 4. Exceptions were some short duration unstressed exposures and the compression creep exposure check test which were exposed in the circulating-air oven.

#### Creep Furnace Exposures:

Specimen assemblies were installed in the furnace and heated to the desired temperature as rapidly as possible, using caution to avoid temperature overshooting of any of the specimens. Specimens were then loaded by gently adding weights to the weight pan. The time for heat-up, from start of heating to start of loading averaged one-half hour (the creep furnaces were limited in rate at which specimens could be heated with accuracy, especially when utilizing a large proportion of the furnace length).

Thermocouple temperatures were recorded continuously throughout all exposures on multi-channel recorders and checked periodically with a precision potentiometer (the latter readings also recorded). Variation in temperature of a given point throughout the exposure period was within  $\pm$  5°F from the average temperature, after the first hour of exposure. Usually, for the first hour temperature was less stable, within about  $\pm$  7.5°F (excepting the 300°F-1 hr exposures, discussed further below). Temperature variation during exposure over specimen 2 inch gage lengths was within 3.5°F. The temperature records obtained

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where used to determine specimen average temperatures for the entire exposure period. These are the actual exposure temperatures recorded in Tables of Results along with nominal exposure temperatures. Actual temperatures were up to  $\pm 4.0^{\circ}F$  from the nominal exposure temperatures.

At the conclusion of the exposure period specimens were unloaded, removed from the furnace, disassembled from all linkage, and allowed to cool on a wood surface.

Procedures for each of the exposures in sequential exposure tests were the same as above except that specimens were usually transferred to a different creep furnace for each part of the sequence. This was done because it was found that rapid heat-up could be obtained more accurately by starting with a cool furnace.

An analysis was made of effect of heat-up time for the various exposure conditions. The conclusion was that the only exposure condition in which this became important was the  $300^{\circ}F$ -1 hr condition. The check test on the effect of compressive creep exposure  $(375^{\circ}F-3 hr)$  was also short enough to have been in this category, but these exposures were performed in a circulating air oven which provided more rapid heat-up. For the short exposures in the sequential exposure program heat-up times were incorporated approximately into the calculation of total exposure for the sequence ( $\Theta$  value) and so were negligible in the effect on the net accuracy of the sequence.

The  $300^{\circ}$ F-1 hr exposures were given special attention and, for the tests that were reported, heat-up times were held to 15 minutes, with good temperature stability achieved throughout the exposure period. This heat-up time amounts to a possible error in exposure value of plus 50 units of  $\theta_{17}$ . The appearance of the strength reduction curves (Figure 17) at this exposure indicates that this error is negligible as to effect on strength level. Similarly, this error has a negligible effect on the determination of the effects of stress during exposure. This can best be seen in Figure 16.

### Circulating-Air Oven Exposures:

A few unstressed exposures were performed in the circulating air oven used for elevated temperature strength tests, shown in Figure 6. These were relatively short exposures, from 3 to 10 hours in duration. The specimens for each exposure were bound together and a thermocouple was attached to each end of the group. The heat-up time was 15 minutes or less except for the  $600^{\circ}$ F-8 hr exposure; one half hr was required for  $600^{\circ}$ F stabilization. Thermocouple temperatures were determined periodically with a precision potentioneter and recorded by hand.

### Compressive Creep Exposure:

The compressive creep exposure check test followed procedures identical with the elevated temperature compression tests. The same specimen was used, supported in the compression test fixture while subjected to a predetermined compressive stress. With each of these were exposed one unstressed compression and one unstressed tension specimen as controls. Compressive creep exposure specimens were undistorted after stressed exposure and no machining was required between exposure and compression test.

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### Measurement of Greep Strain

Creep strain was measured by using two sets of gage points of the type shown in Figure 5. One had about 2" separation and the other slightly less. The larger set was impressed in the reduced section of the specimens before exposure and the smaller set was impressed within the first set after exposure. The separation between the resulting points at each end of the gage length was measured under a microscope with a Filor eyepiece, at a magnification of 100 power. The sum of the two measurements was equal to the total inelastic strain in the 2 inch gage length plus the original difference in lengths of the two sets of points.

To calibrate this system, it was necessary to know precisely the differences in lengths of the two sets of gage points. To accomplish this, each time a group of specimens for exposure was marked, trial impressions were made on a dummy strip. The differences between sets of points on the test strip was measured precisely under a microscope as before. Variation in creep strain measurement was checked and found to be within approximately + 0.02% strain for the single exposure measurements.

For the long creep exposure specimens, the same procedure was followed in each of the three locations from which a compression specimen was to be machined.

For specimens subjected to multiple exposures, total creep strain accumulated was determined after each exposure step. This was done with succeeding sets of gage marks laterally displaced (up to 0.035 in.) from the original set. It was found that accuracy of creep strain measurement was reduced by the lateral displacement of gage marks. Performance checks indicated that maximum variation of measured sequential exposure strain from the actual was about 0.06%.

### Specimen Processing Between Exposure and Strength Testing

### Tensile Specimens:

For tensile specimens, steps between exposure and tensile testing consisted of: creep strain measurement, hardness testing (if applicable, see below), and holding at room temperature for a standard delay time. The first two items are discussed elsewhere under test procedures. The standard delay time is explained as follows: Up to 1 1/2 hours were required to complete tensile tests of some groups of specimens exposed together. To minimize differences between specimens as to holding time at room temperature before testing, it was decided to delay start of testing for a standard time of one hour after the end of the exposure period. This step was taken against the possibility of minor changes in strength due to secondary ageing reactions at room temperature after exposure to temperature.

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### Compression Specimens:

Steps between exposure and compression testing consisted of:

Creep strain measurement Cutting compression test blanks from reduced section of creep exposure specimens Hardness testing (if applicable) Refrigeration storage of compression test blanks Machining of compression specimens from blanks

All of the above stops are explained elsewhere or are self explanatory except the fourth. Refrigeration of compression test blanks was used to avoid long and variable soaking times at room temperature caused by the intermediate machining step. Blanks were refrigerated to below 5°F within an hour after the end of exposure and held from a minimum of 12 hours to a maximum of four weeks before bringing back to room temperature for machining and compression testing. Time at room temperature for machining and compression testing varied from 3 to 8 hours. Total time at room temperature between exposure and compression testing was therefore from 4 to 9 hours for all specimens.

The effect of refrigeration was checked on unexposed control specimens for room temperature test. Two such specimens were refrigerated for about 16 hours while several specimens located near these in the sheet were not. All were compression tested at room temperature. No effect of refrigeration could be discerned in the results (Table 3).

### Hardness Testing

All specimens for room temperature strength tests were hardness tested after exposure and before strength testing. Two Rockwell hardness scales were used because of the wide range of hardness of specimens after exposure, the B scale and the H scale. Most specimens were tested on both. Location of hardness tests on tensile specimens was at each end of the reduced section. Compression test blanks were hardness-tested at each end on material subsequently removed in machining of compression specimens.

A number of hardness test results on creep-exposed tensile specimens have been omitted from the results reported. These tests were performed just outside the specimen reduced section, through crror, on material which had been subjected to a lower exposure stress than that in the reduced section. These values would not have been completely representative of the material which was subsequently strength tested and so were deleted.

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Strength Testing

Tensile Tests:

After exposure, tensile specimens were tested at one of four test temperatures, room temperature, 200°F, 300°F or 400°F, using the equipment shown in Figure 6 for the elevated temperature tensile tests and most of those at room temperature. Properties determined were ultimate tensile strength, 0.2% offset yield strength, and percent elongation. Yield strength values were obtained from autographic stress-strain curves using extensometers over a two inch gage length. Percent elongation was measured over a two inch gage length with dividers.

Specimen temperatures were measured by a thermocouple contacting the specimen edge at the midpoint between ends of the reduced section. In tests to verify this procedure, there was no difference between readings of thermocouples located in holes drilled in the specimen and those at the edge location, nor was there any significant temperature difference from top to bottom of the reduced section. Temperature was read from a precision potentiometer. Readings were recorded at the start of loading and at maximum load. The actual test temperatures given in the tables of results are the averages of these two values, which were never more than 3°F apart.

The maximum time for heating specimens to test temperature was 12 minutes and this was for the 400°F test temperature. Less heat-up time was required for the 200°F and 300°F test temperatures, but a standard heating time of 12 minutes before loading was also used for these. Times from start of loading to the 0.2% offset yield load and to maximum load were obtained on elevated temperature tests, using a stopwatch. Time to yield varied between 20 seconds and one minute. Time from start of loading to ultimate load varied between 45 seconds and 3 1/2 minutes. (At 300°F and 400°F test temperatures maximum time was 2 1/2 minutes.) An exception to this were tests on the nearly annealed material (600°F-8 hr exposures). Some of these required 5 minutes to reach maximum load.

All tension tests were performed with a constant rate of crosshead travel from start of loading to fracture. A few of the first room temperature tensile tests were performed on a Baldwin SR-4 test machine at a strain rate of 0.005 in/in/min up to yield, with the aid of strain pacing. Shortly thereafter the Instron Machine, with a circulating air furnace (Figure 6) became available, and strength testing was changed over to this machine. The strain rate up to yield for tests conducted on the Instron Machine averaged 0.01 in/in/min. Data provided in Reference 11 indicate that a change in strain rate of this magnitude should not significantly affect the results of tensile tests on 7075-T6 at room temperature.

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Strain rates up to yield were calculated from readings of time to reach the 0.2% offset yield load. These values varied from the average value quoted above, depending on the strength level of the material and the test temperature. All strain rates fell within 0.008 and 0.012 in/in/min. This degree of variation would be undesirable for tests at 400°F on this alloy; however, nearly all of the 400°F tests fell within 0.008 and 0.01 in/in/min. The few that did not were tests of material in the nearly annealed condition. Actual strain rates up to yield for tests at 400°F are given in the tables of results (see Tables III and XVII).

Strain rate in the specimen reduced section tended to increase after yielding, the usual case when crosshead travel rate is held constant throughout tensile tests. (With specimen yielding the change in stress rate causes a change in distribution of strain between the specimen reduced section and other parts of the linkage between the test machine crossheads.) Actual strain rates from yield to ultimate were not obtained, however all fell between approximately 0.01 in/in/min and 0.05 in/in/min. The latter is the maximum possible rate with the 0.1 in/min crosshead separation rate used. Also, specimens of approximately equal strengths tested at the same test temperature would have had comparable strain rates from yield to ultimate.

#### Compression Tests:

After single exposures compression specimens were tested at one of four test temperatures, room temperature, 200°F, 300°F or 400°F; and after multiple exposures at room temperature or 300°F. 0.2% offset compressive yield values were obtained from autographic stress-strain curves. Test facilities were the same as for tension tests.

The fixture for supporting compression specimens during tests is shown in detail in Figure 9 and in place in the test machine in Figure 10. Specimens were supported by guide plates having offset vertical grooves. Specimen alignment in the fixture and tightening of guides were by "feel". Strain was measured by a two inch averaging extensometer, using extension arms for elevated temperature tests. Accuracy of compression test procedures was checked by inspection of modulus of elasticity values from a number of compressive and tensile stress-strain curves obtained at room temperature and 300°F.

Specimen temperatures were measured by a thermocouple attached to the edge of one of the specimen guides, at the midpoint between specimen ends, about 0.125 in. from the specimen surface. The test furnace and fixtures were stabilized at the test temperature before installing compression test specimens.

Maximum time for specimens to reach any of the test temperatures was slightly under 12 minutes, sc heating time was standardized at 12 minutes for all elevated temperature compression tests. This agreed with the heating time for elevated temperature tension tests. Maximum time from start of loading to the 0.2% offset compressive yield load was 1.2 minutes.

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The accuracy of compression specimen heating was checked with a dummy specimen which had thermocouples located in grooves at the top, middle and bottom in addition to the thermocouple on the specimen guide. No significant temperature disagreement occurred between any of the thermocouples when procedures which simulated compression tests were followed.

Compression tests were performed with a constant rate of crosshead travel. Strain rates as determined by stopwatch readings averaged 0.009 in/in/min for all tests. Individual values obtained varied between 0.007 and 0.001 in/in/min with a few exceptions. Part of this variation is probably due to the method of measuring strain rate. All tests at 400°F fell between 0.0075 and 0.0095 in/in/min. Strain rates for 400°F tests and for the exceptions noted above are recorded with the test results (Tables III, XI and XVI).

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Equipment used in the experimental portion of this investigation can be conveniently divided into two categories.

Exposure Equipment: Equipment used to provide the desired exposure conditions for specimens prior to testing consisted of:

- . Creep-rupture machines
- . Temperature recording equipment
- . Creep strain measurement equipment

Test Equipment: Equipment used for tensile, compressive and hardness testing of specimens after exposure consisted of:

- . Tension and compression test equipment
- . Compression test fixture
- . Load-strain recording equipment
- Hardness tester

### Creep-Rupture Machines

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Stressed and unstressed exposure of specimens was carried out on several creep rupture machines. Six of these on which most of the exposures were performed were of the type shown in Figure 4. These are Arcweld M-3 creep frames with cylindrical wire-wound resistance furnaces, 32 inches long by 2.5 inches inside diameter. Specimen loading is by weights acting on a lever. Furnaces are controlled by Wheelco 407 current proportioning, indicating controller in conjunction with a chromel-alumel thermocouple mounted on a specimen in the center of the furnace.

Three Arcweld XJ creep frames were also used. These had 22 inch long by 4.5 inch inside diameter wire-wound resistance furnaces. Function of the XJ units is similar to the M-3 units previously discussed. Furnace control is by Brown Electronik electropulse proportioning recording controller.

# Temperature Recording Equipment

Specimen exposure temperatures were continuously recorded on one of two Brown Electronik 12-channel recorders. These recorders have a rated accuracy of 0.25 percent of full scale temperature range. The scale used was zero to  $1200^{\circ}$ F. Specimen temperatures were periodically checked with a Leeds and Northrup model 8662 precision potentiometer. Thermocouples used throughout all tests were made of special close tolerance 24 gage chromel and alumel wires joined by flash welding. Thermocouples were within  $\pm 1^{\circ}$ F accuracy in the temperature range of usage.

### Creep Strain Measurement Equipment

Apparatus for measurement of creep strain consisted of two sets of gage points, one of approximately 2 inch separation, the other slightly less than

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2 inches. One of these is shown in Figure 5. The points used in these gage marking devices are Rockwell Diamond Point Indenters. Actual measurement of the separation of the two sets of points at each end of the 2 inch gage length was accomplished with a Bausch & Lomb bench microscope with Filor eyepiece at a magnification of 100 or 125 power.

### Tension and Compression Test Equipment

Two test machines were utilized for tension tests. A few of the first room temperature tensile tests were performed on a Baldwin SR-4 hydraulic universal testing machine, with a 50,000 pound maximum capacity. This machine is equipped with a Baldwin strain pacer operated from the extensometer differential transformer. The remainder of the tension tests and all compression tests were performed on an Instron TTC-ML test machine, shown in Figure 6. Tests on the Instron machine were run at a constant rate of crosshead travel. A stopwatch was used to measure time from start of loading to the .2 percent offset yield load and maximum load. For elevated temperature tests, a circulating air electric-resistance oven manufactured by Missimers Inc. was employed (shown in Figures 6 and 7). This oven has a working chamber 15 inches wide by 12 inches high and 22 inches deep.

Furnace temperature was controlled by a Brown Electronik electropulse proportioning recording controller, in conjunction with a chromel-alumel thermocouple located in the furnace working zone.

## Compression Test Fixture

The compression test fixture consisted of a pair of adjustable specimen guides, a loading system and a base for the specimen. Details of the fixture are shown in Figure 9. The compression test assembly is shown in Figure 10.

In this fixture, specimen guides have vertical grooves offset from each other. This has been indicated as a generally acceptable specimen supporting arrangement in Reference 9. The specimen is loaded at the top by an unattached blade which is supported by the guides and protrudes slightly at the top of the guides, where it contacts a loading ram. The blade thickness is a slight undersize of the specimen to avoid binding in the guides. Load is iransmitted through a 5/8 inch diameter ball in the ram assembly to assist in alignment of loading.

Material used in the frame, guides and loading ram is A-286 alloy, with Stellite No. 1 hard facing in highly loaded areas. The loading blade is hardened tool steel. The ball is a standard ball bearing which is replaced upon any sign of plastic deformation.

### Load-Strain Recording Equipment

A Baldwin P3-5M microformer extensometer was used to measure strain in all tensile tests. This extensometer is an averaging, separable type. For tests at room temperature, 200°F and 300°F, it was attached directly to the specimen with opposing conical points, over a 2 inch gage length. For tension tests at 400°F extension arms from a different extensometer were used with the PS-5M. These were also attached to the specimen with opposing conical points

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over a 2 inch gage length, with the arms extending outside the furnace. The installation for tests up to  $300^{\circ}$ F is shown in Figure 7, and for  $400^{\circ}$ F tests in Figure 8.

Inring compression tests strain was measured with a Baldwin SRIE resistance wire strain gage type compressometer. This is also a strain averaging instrument, and was used over a 2 inch gage length. For tests at room temperature the compressometer was attached directly to specimen edges by spring loaded knife edges. For elevated temperature tests, a set of extension arms were used which gripped the specimen edges in a similar fashion and extended outside the furnace. The SRIE was actuated by the relative motion of the extension arms, unmagnified by any lever system. The elevated temperature compression test assembly is shown in Figure 10.

Load-strain curves were obtained on the Instron machine X-Y recorder. Load was recorded on the Y axis as measured by the test machine SR-4 load cell. Strain was recorded on the X axis either from the extensometer differential transformer signal or the compressometer signal after conversion by an SR-4 converter.

### Hardness Tester

A Rockwell hardness tester, Wilson model 4JR was used with 1/16 inch diameter and 1/8 inch diameter indenters to obtain Rockwell B and H scale hardness readings. Hardness test accuracy was periodically checked by hardness tests on calibrated test blocks.

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# TABLE 1SINGLE EXPOSURE CONDITIONS TESTED

	EXPOS	URE CONI	DITIONS		FINAL TEST	TEMPERATURE
H17 Exposure	Temp.	Time	Stress (tensil.)	Target Strain	Tension	Compression
×10-3	٥F	hrs	ksi	<u>×</u>	o <sub>F</sub>	o <sub>F</sub> .
12.92	300	1	υ	0	R.T., 300	R.T.300
12.92	300	1	50	1.0	R.T., 300	R.T.,300
13.49	250	100	0	0	R.T.,300,400	R.T.,300,400
13.49	250	100	35-36	0.1	R.T.,300,400	R.T.,300
13.49	250	100	<b>►.</b> {	1.0	R.T., 300, 400	R.T.,300
14.44	300	100	Э	0	R.T.,200,300,400	R.T.,300
14.44	300	1.00	20	0.1	R.T. 300	R.T.
14.44	300	160	32	0.3	R.T. 200.300.400	RaTa
14.44	300	100	34-35	1.0	R.T.,200,300,400	R.T.,300
14.58	35()	10	0	0	R.T. 300	R.T. 200 300 400
11.58	350	10	20	ŏı	и <b></b>	P T 200
14.50	250	10	20	0.2	иото, 200 рат. 200	D m 200
14.58	350	10	20-22	10	РТ 300	R T 300 100
14.90	550	10	20-22	1.0	R+1+, 300	R+1+3500,400
14.89	350	24	0	0	R.T., 300	
14.89	350	24	24	0.3	R.T.,300	
14.89	350	24	27	1.0	R.T.,300	
15.31	350	80	0	0	R.T., 300	R.T.,300
15.31	350	80	13	0.1	R.T., 300	R.T., 300
15.31	350	80	19	0.3	R.T.,300	R.T., 300
15.31	350	80	21	1.0	R.T.,300	R.T., 300
15.48	400	10	0	0	R.T. 200.300.400	R.T. 200.300.400
15.48	1:00	10	13	0.1	R.T. 200, 300, 400	R.T. 300
15.48	400	10	17	0.3	R.T. 200, 300, 400	R.T. 300
15.48	400	10	19-20	1.0	R.T.,200,300,400	R.T., 300, 400
16.38	450	10	0	0	R.T. 202.300.400	R.T. 200.300.400
16.38	1.50	10	10	č.1		R.'f.
16.38	1.50	10	13	1.0	8.1.200.300	R.T. 300
	4,70	10	-7			
17.28	500	10	0	0	R.Y. 300	R.T.,200,300,400
17.28	500	10	9	1.0	R.T., 300	R.T., 300
18 <b>.98</b>	600	e	0	0	R.T., 200, 300, 400	
		(Cc	mpressive)			
14.59	375	3	0	υ	R.T.	R.T.
14.59	37 <b>5</b>	3	28	-0.3		н.Т.

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	SEQUENCE I			EXFO:	SURE CO	NDITIONS		FINAL TEST	TEXPERATURES
Code	Description	Total $\theta_{1''}$ Exposure x10-3	Step No.	Temp.	Time hrs	Stress (tensile) kai	Target Strain	Tension	Compression Or
								• 	· r
A-1	Decreasing temp. trend,	•	1	350	3.5	31.0	•		
	ureu ačiesa	11.6	- <b>∡</b> 3	313	100	32.5	1.0	J # 300 (00	
		1400	ر	41)	100		1.0	R.1., 300,400	R.T., 300
A-2	Same as A-1 sequence ex	cept lower	exposu	re stres:	30.9.		0.25	K.T., 300, 400	R.T.,300
A'-1	Inverse of A-1.	•	1	275	100	34.5	•		
	Increasing temp. trend,	•	2	315	20	32.5			
	high stress	14.6	3	350	3.5	31.5	1.0	R.T.	
A1-2	Same as A*-1 sequence e:	xcept lower	· expos	ure stre	5365.		0.25	R.T.	
D. 10	The stops of P-14		1	300	1641	29.5			
0-12	two scens or p-tr	15.0	2	350	21.	21.0	0.50	R.T., 300,400	
					100				
B-13	Three steps of B-14	•	1	300	100	29.5	•		
	Increasing temp. trend.	15.3	2	100	3.5	19.0	0.75	R.T. 300.400	R.T. 300
		1,,	,			-///			
B*-13	Inverse of B-13	•	1	400	3.5	21.0	•		
	Decreasing temp. trend.	•	2	350	24	20.0	•		
		15.3	3	300	100	19.0	0.75	R.T., 300	
8-17	Mixed temp, trans.		1	300	160	29.5	•		
0 - 14	high stress	•	2	350	24	21.0	•		
		•	3	00	3.5	19.0	•		
		15.4	1.	150	21.	20.0	1.0	R.T., 300,400	R.T.,300
B-24	Same as B-14 sequence es	cept lower	. exhoar	urc stre	iaca•		0.25	R.T.,300	R.T.
			,		100				
8-1Ja	Modilled D-13	•	2	150	26	22-0			
	Zelo-actest Tast. Steb	15.3	3	1,00	3.5	Ű	0.50	K.T.	
8-14 <b>a</b>	Modified B-14	•	1	300	100	29.5	•		
	zero-stress last step	•	2	350	24	21.0	•		
		15.4	у 4	350	24	0	0.75	R.T., 300	R.T.
						-		•	
D	Zero-stress first step	•••	1	350	72	0			
	High stress last step	15.3	2	375	3.5	25.0	1.00	K.T., 300	R.T.
E-1	Complex 10 part mixed	•	1	350	10	210	•		
	sequence	•	2	325	40	0	•		
		•	ן ג	300	2+2	««+) 21.,0	•		
		•	45	350	10	22.0	•		
		•	6	325	40	19.0	•		
			7	375	3.5	õ	•		
		•	8	300	100	21.5	•		
		•	9	325	40	16.0	•		
		15.4	10	375	3.5	0	0.80	R.T.,300	R.T.
E-2	Same as E-1 sequence exc	ept lower	exposur	e stress	44		0.25	K.T.	
				-			0.14	5 4	
Ë-3	Same as E-1 and E-2 sequ	ences exce	be <b>tom</b> e	ar expos	nie sri		A*13	Re i e	

# TABLE II SEQUENTIAL EXPOSURE CONDITIONS TESTED -- STRESSED EXPOSURE<sup>(1)</sup>

(1) Note: An equivalent unstressed sequential exposure program was provided by zero stress control specimene included with each exposure in this table.

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TABLE III TENSILE AND COMPRESSIVE STRENGTHS -- UNEXPOSED

							٩	PENSION	ı						COMPAR	3310
IDE	NTIPI(	CATION	HARDNE	59		PROPERT	TES AT		CORNE PROPERT	LTED IES (1)	) ta	DIT171	ICAT ION	PROFZ. TEXP	ATTES AT	CONRECTED PROPERTIES (1)
Funiral Test Test	1 Te 81	t igu*.παr • <sup>+</sup> : a	iachrae Nar Ir H	11 89	Artua Inst Tunji	1 F.y	Tr <sub>ea</sub>	ator tr. 21	y	P.,,	Test Test Temp	Test Po.	Specieer So.	Attus Test Test	1 <sup>1</sup> cy	r <sub>c7</sub>
			1491	•1,1		1.1	ieų ·	*	put	F#1				~	te3	pei
a.T.	11	Atue			eo		77.XC	) 11.0		78,100	н.т.	~11	CIICT()	) જા	73,000	73,300
k.T.	71	A210			- 3	- 7,200 - 7,100	0 18,500	11.5	- 07,200 - 08,100	78,500	, K.T. ) H.T.	211	- CLICH - CLICH(2)	91 ) 91	72,300	72,600
H.T.	21	A211			15	47,900	76,UCU	11.5	61,900	78,000	T.N.	211	C350T	82	75.500	75,700
н.Т.		12.0			20	07,200	77,300	16.5	67,500	17,500	) R.T.	211	C; 308(2)	82	75,400	75,600
N.I.	71	A310			60	67,700	78,100	11.5	000,90	76,300	2		-	-	A-TRC+	71.,300
H.T.	71	A350			80	67,300	77, 100	10.5	67,000	77,500	, ;					
k.T.	71	AL.10			60 10	17,200	71,800	11.0	67,500	78,000						
H.T.	21	AU.50			+0	67,400	77,400	) 13.0	67,700	77,600	) )					
R.T.	71	A520			70	49,200	77,000	14.0	69,200	77,800	2					
R.T.	2 2	A560			47 147	70,200	80,000	12.5	70,000	80,500	) }					
R.T.	71	4620			67	69,700	78,600	11.5	69,500	78,500						
л.г. R.T.	71	AGLO A(.60			70	70,100	79.300	13.0	70,100	77,300	•					
A.T.	- 71	A710			CO.	68,500	78,900	11.5	68,800	79,100	)					
R.T. 8.T.	71.	A730 A750			eo eo	68,000	78,700	10.5	68,300	78,900	•					
A.T.	n	AB10			80	68,900	79,700	11.0	69,200	79,900						
R.T. R.T.	71	A830			80 80	69,100	78,700	11.0	67,900	78,900						
R.T.	71	A920			75	67,700	77,400	11.5	67,700	77,100						
R.T. R.T.	71 71	A940 A941			75 75	65,800	76,700	11.5	65,800	76,700	•					
R.T.	'n	4960			75	68,000	78,900	10.5	68,000	78,900						
R.T.	71 71	A1020 A1060			75 75	67,100 68.100	77,700	10.0	67,100	77,700						
A.T.	n	A1110			80	64,800	78,300	12.5	69,100	78,500	l					
n.T.	71	A1130			80	69,100	76,300	10.5	69,400	78,500						
R.T.	'n	A1210		${\bf t}_{i} \in {\bf t}_{i}$	ŝč	68,700	78,200	10.5	69,000	78,400						
R.T.	71	A1230			80	69,100	78,600	12.0	69,400	78,800						
R.T.	102	A1240(4)			75	70,800	77,700	10.7	70,800 68,300	77,700 78,300						
R.T.	120	B131	115.5 6	0.0	74	66,800	75,600	10.5	66,800	75,600						
R.T. A.T.	102	B166 (1.)	110.0 6	0,0	77	71,200	61,100	11.5	71,200	#1.100						
R.T.	120	8320	116.0 8	1.0	74	66,500	76,200	11.0	66,500	76,200						
ж.т. З.Т.	120	B100 B1266(L)	117.0 #	<b>7.0</b>	77	72,900	10,500	10.5	87,000 72,900	76,700						
						10 000	Average		67,000	76,400						
R.T. R.T.	211	C210 (5)			73 91	68,000	76,400	14.0	68,500	76,300						
R.T.	211	C371 (5)			82	68,200	76,300	13.5	68,500	76,600						
R.T.	128	6636			73	67,900	75,100 Average	12.5	68,400	75,100						
	-				<b>~</b> ~		40 000		4						/ /	
200	1	AJ33 A431			200	65,000	69,700	13.0	65,000	69,700	200	223	C725-2	202	72,500	9,200 12.600
200	71	A#20		:	200	65,100	70,400	12.0	65,100	10,400	200	22)	C725-3	203	71,700	1,600
							YAALUEA		05,000	99,900	200	203	(72)-10	303	au,ouo e lverage 7	7,100 10,700
200	71	4130			-	47 600		14.0	67 400 0	14 200	***	••••		-	43 mm 4	
100	71	A256		5	02	56,700	57,600	20.0	56,900	7,000	300	22)	C725-5	300	63,400 a	3,600
300	72	A720		2	296	56,900	50,500	21.0	56,700	58,300	300	223	C725-6	300	62,000	2,000
300	120	8230		4	299	54,300	56,400	10.5	54,200	6, 300				-		, , , , , , , , , , , , , , , , , , ,
300	120	B4,20 B4,60		2	299 100	54,400	56,100	18.0	54,300	14,000						
							AVOTAR		x, 100	.,200						
300	224	C7217		1	100 102	56,500	58,300	16.5	56,500	18,300						
	~~'1				~~	,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,	AVERAGE	- 7/0	56,600 5	A,100						
600	70	1217 (1)		4	05	41. <b>80</b> 0 (	1.000	14.0	12 100 1	1.700		221	(1)(1)	100	-	
.00	n	1266 (3)		3	90	4,000	5,000	11.5	43,700 4	4,000	400	22)	(725-0(1)	400	u, 300 v	6,900
400	71	A610(3)		1	97	42,200	13, 100	15.5	11,900 4	3,100	400	22)	C725-9(3)	100 1	UN,700 V	700 7.900
100	234	11212 (3)		4	01	12,700	1,600	13.5	12,000 1	3,700						
1,00 1.00	274	14,25 ( ) } N.C.L. 1		4	ກເ ຄາ	12,700 I	11.00 11.100	16.5 11.4	12,100 L	4,500						
							Averar!		12,700 1	1,.00						
400) 100	27%, 271	(7221(1))		4	172 i 19 <b>9</b> i	11,200 l	4,300	10.0 17.0	47,543) 4 12.330 1	4,(410 6,3(2)						
				,			VC PWP-		1,100 4	6,400						

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Properties corrected to mainal test temperature.
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 Break refrigerated from sorrage, about secretance tests taken from about correct.
 Tonelle test reformed on a croop specimen. (See Figure 2)

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	E.	Rteld	C.974		0°997 0°586		0.893 0.876 0.892			
		Fra Fat	71,100	77,100	70,700	007 LL 007 LL 007 LL	73 560 73 560 73 500	222888 222888 222888	r:35333 855333	68.600
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		relog Star	0.2 2.5	0.51 2.52	201	**** ****	10.5		000	•
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TABLE IV (Continued) ROOM TEMP. TENSILE STRENGTH AFTER SINGLE EXPOSURES

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TABLE IV (Continued) ROOM TEMP. TENSULE STRENGTE AFTER SINGLE EXPOSURES

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<b>37,000</b> 16.5 15,550 37,500 0.2287) <b>36,700 18.0</b> 15,800 25,700 0.227 280 <b>36,600</b> 16.9 15,200 25,500 0.22142
<b>37,000</b> 16.5 15,500 37,000 0.228 <b>36,700 18.0</b> 15,200 55,700 0.228 <b>36,600 18.0</b> 15,200 55,700 0.227
<b>37,000</b> 16.5 15,550 37,000 <b>36,700</b> 18.0 15,200 55,700 <b>36,600</b> 16.5 15,200 55,700
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TABLE V 200<sup>0</sup>P TENSILE STRENGTE AFTER SINGLE EXPOSURES

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(1) Properties corrected to matinal appears conditions and sominal test temperature.

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		ries(1)	Ftu Fs:	36 <b>,</b> 73	59,900 59,900 59,200	21,200	seraran 588588	22,50 25,50 35,50	20000000000000000000000000000000000000	1,200 11,700 19,200	41,288 41,2888 41,2888 41,2888 41,2888 41,2888 41,2888 41,2888 41,2888 41,2888
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TABLE VI (Continued) 300<sup>0</sup>F TENSILE STRENGTE AFTER SINGLE EXPOSURE

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TABLE VI (Contineed) 300<sup>9</sup>F TENSILE STRENGTE AFTER SDNGLE EXPOSURE

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TABLE VII 400°F TENSILE STRENGTH A FTER SINGLE EXPOSURES

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TABLE VI (Cectourd) TENSOLE STRENGTH AFTER SI

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TABLE VIII ROOM TEMP. COMPRESSIVE STRENGTH AFTER SINGLE EXPOSURE

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(1) Properties corrected to meeting, exposure conditions and nominal test temperature.

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TARLE VIII (Continued) ROOM TEMP. COMPRESSIVE STRENGTH AFTER SINGLE EXPOSURE

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	•	35 25 25 25 25 25	<u>88</u>	*****		**********	88888 <u>8</u> 8	( mit )
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nell Iress		75.5 78.5 78.5	75.5	1222222222 000220220 0002000	63.0 61.0 62.5 62.0 61.0 61.0	8888888554 99999999999999999999999999999	8888888 \$998888	
Actual Test Temp.	сŁ.	5555	12 ki	******	228825 288255	rrr <b>a</b> garry	££££ <b>£</b> £	
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MOUM ILMP. COMPRESSIVE STRENGTH AFTER SINGLE EXPOSURE

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.'έ.										6.375 1975	1950							c.263					0.800	0-780
Fcy	181	-1,100							32.5	21,900 28,300	28,30 28,30 20,30	200	29,200	28,400	27,100	25,900	8	200 200 11	19,400	19,200			59,600 59,600	57,900 56,100
Fcy	psi	00°00 10°300			35				3.135	21,900 25,200	26,200 28,200	200	29,200	28,400	27 - 20 27 - 20	20,800		38 38 3	19,400	19,200			59,000 59,100	57,500 56,300
Actual Test Temt.	°F	۲. ۲ ۲. ۲	2 2	ĩ	01		0	<u></u>	i fi	ដ	<u>.</u>	````	វដ	ដ	n Yr	5	;	v.Č	56	ξ- <u>ξ</u>	<u>,</u>		ጵያ	<u>ጽ</u> ጽ
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Total Strair	N	18 0	58	5	0.20	72-0	<b>7.</b> 0	0	0.70 C.95	01	000			23	10	C.68	1	00	1.15	2.2			00	0 .32
Time	ይ	29	2			2	2	ទ	13	25	335	4	22	12	20	10		22	01	22	2		<b>س</b> س	<b>m</b> m
Temp.	о М	ជីទ	38	3	ថ្មី	8	8	<u>8</u>	10 10 10	-25	클릭신		39	3	537	121		នន	ĝ	(g §	ł		375	378
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Total		1.0	, , , , ,	1.0	0.35	C. 3	50.0	0 i	00	0.	יטנ	,	33	6-1		0-1		υu	1.0		2	:121	00	ဝမို
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		15-18		15.46	15.45	15-48	158	15-48	15E	3C•35	15-36 15-36		1e.36	10.36	16.36	16.36		22.5		2		SURE STRE	14.63 [][	69-7.
	h	24 S	21		0	2	X	33	o u H M	51			22	13	25	12		អង		12	4		<b>6</b> 7.6	\ <b>r</b> \ <b>r</b>
	c,F	8	32	8	8,	ų,	8	នុះ	នុទុ	×:	4.2.1. 7.2.1	2	4	4 <i>4</i> ,	9 9 9			8. <b>X</b>	ÿ	(8)	X	2017 IES	51	1 E

TABLE IX 200°F COMPRESSIVE STRENGTE AFTER SINGLE EXPOSURES

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RATICE	ين بر				
TENERITE	<b>L</b> a:	C.828 C.801 C.757	C. 550 C. 555 C. 543	0.371 c.370 c.363	c.329 c.292
LAKEUTAD ROPERTIES	F <sub>cy</sub> psi	59,500 59,500 57,000	40,900 41,200 40,400	27, 500 27, 500 27, 000	23,700 21,700
	F <sub>cy</sub> psi	0,900 53,900 55,400	-0,900 -1,200 10,100	27,600 27,500 27,000	23,700 21,700
PROPERTI AFTER ELPOS	Actual Test Temp. F	202 201 202	888 888 888	888	201 203
	Specimen Group Ko.	สลัส	สสส	236 236 236	<b>R R</b>
SMOITICH	Total Strain	000	000	000	00
PO. ARE CC		999	222	999	10
ACTUAL EX	1 1 1 1 1 1 1	35 32 32	888 888	151 151	ĕĕ
	Specimen No.		CI167 CI168 CI168	C 5337 C 533N C 533B	C626N C626B
'n	Beral Strain	000	000	000	00
DOFT TOK	81.24 71.75	000	000	000	00
	(17+10-1) x10-3	888 777	15-48 25-48 15-48	16 <b>.3</b> 8 16 <b>.3</b> 8 16 <b>.3</b> 8	17.26
TOPCINT 1	1	999	200	929	10
	÷ •	***	888	333	88
WADC TH	56-58	5 Pt II			51

(1) Properties corrected to nominal exponence conditions and nominal test temperature.

TABLE X 300<sup>0</sup>F COMPRESSIVE STRENGTH AFTER SINGLE EXPOSURE

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SELLER ON	ະກີ່ເຊ ເວັ້ອ ເກີ່ອີ	e1,100 e1,100	59,200 57,500 57,400		57, 800 56, 500 56, 500 57, 500 57, 500	50,100 56,500 49,700 49,100	77,400
SURE F	Fcy psi	61,100 56,900	57,200 57,400 57,400	11-11-12-86 96-11-12-86 96-11-12-96 96-12-96 96-11-12-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-10-96 96-100-96 96-100-96 96-100-96 96-100-96	56,300 56,100 56,100 56,100 57,000	55, F00 56, 300 49,400	4/200
AFTER EOPC	Actual TestoFemp.	302 302	858	555 <b>89898</b> 8	5555555 5555555	ដ្តី <b>ខ្លីខ្ល</b> ី	<b>4</b> 47
	Specimen Group No.	228 226	228 228 228	217 217 200 200 200	240 240 217 217 217	218 218 218 218	212
SNOTTONS	Total Strain	ن ن	0.86 1.22 2.01	603000	10000000	111 00	4.1
POSINE OC	Time hr		ннн	883888	888888	88 888 888	3
ACTUAL E		2 <del>38</del> 299	888	XXXXXXXX	2222222	ត្ថ ខ្លួត	The
	Spectmen No.	C)108 C6168	C51oT C516M C516B	い 1000 1	C6311 C6311 C63118 C63118 C5354 C5354 C5354	C6258 C6258 C6258 C6258 C6218	01700
	Totel Strein S	00	888	00000	01.0 01.0 01.0 01.0	888	
NULTION	Strees krii	00	888	063000	36.0 36.0 5.1 1.1 1.1 1.1 1.1 1.1 1.1 1.1 1.1 1.1	00 % 0.%	2.00
EDPUSIBLE	T(17+10 t	8.11 8.11	222 222	<b>6</b> 6 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7	67-61 67-61 67-61 67-61 67-61	11 11: 11 11:	
TADADA	1 A			853888	3888888	<u>88</u> 888	3
		ăă	ន្តន្តដ	******	****	<u>88 88</u>	ž

(1) Properties corrected to maid and expense conditions and nominal test terperatures.

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TABLE X (Continued) COMPRESSIVE STRENGTE AFTER SUNCLE EXPOSURE

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£ ♪ 6-585	1 x	11.7417	1	Land a	Aperiment	į.		Total Seruin S	Specimen Group Yo.	Achual Test Temp. op	F <sub>cy</sub> Fai	F <sub>cy</sub> psi	ž.	ц, Ц
8888 <b>r</b> 11	3233	8888 1111	0000	0000		y zy y	9999	0000	£855	<mark>୫</mark> େଟ୍ଟେ	55,600 52,200 52,200	55,60 52,500 52,800 51,800	5.728 5.734 0.690 0.690	
2 222	9 999	8 888 1 111	000	0 0,10 0,10		র রষ্ট	01 01 01	0 200 1988	<u> </u>	a 8885	53,700 53,700 53,700	3 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8		1997 1997 1997 1997
<u>8888888</u> 53	2222222	<b>***</b>	<b>***</b> ****	222888		E E E E R A R A R A	2222222	1110-1-0-0 1111-1-0-0 1111-1-0-0	555577	\$ <b>888</b> 888	<b>3,38,</b> 38,88 <b>3,38,</b> 88 8,88 8,88 8,88 8,88 8,88 8,88 8	<b>88.6</b> 1988 1997 1998 1998 1998 1998 1998 1998		4.969 5.966 5.955 5.955 5.919 5.919
<u>828</u> 8	888 8	1111 1111 1111 1111 1111	000	0 0 10		<u> </u>	888 8	000 0 00	67 861 861 861	888 8	38,200 37,700 36,900 37,500	38,200 37,800 37,200 37,510	c. 51. c. 509 c. 501	C. 99.
22222222	8888838	******	99999955 2222225	832222511 83222258		<u>88888888</u>	***	8.00 8.00 8.00 8.00 8.00 8.00 8.00 8.00	5588866	8688558	77 88 88 88 88 88 88 88 88 88 88 88 88 8	37,78 37,58 38,58 38,58 88,588 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,5888 88,58888 88,58888 88,58888 88,588888 88,5888888 88,588888888		0.963 0.999 0.981 0.981 0.981 0.981 0.981
<b>33</b> 3 <b>3</b> 3		22222 22222				<b>33333</b>	00000	00000	888 <b>11</b>	88588	35,200 35,200 35,500 35,700	38,800 56,200 35,400 35,400 35,700	0-522 0-487 0-491 0-491	
<u>999999999</u>	9999999999			999955551 999555888		££88£88££	9999999999999	6.1.6.6.6.6.0 6.1.6.6.6.6.6.6.6.6.6.6.6.6.6.6.6.6.6.6	<b>នំនំនំន</b> ំឱននិនិនី	<b>288888</b> 888888	88868888888 88888888888888888888888888	88888888888 88888888888888888888888888		0.967 0.978 0.978 0.971 0.973 0.073 0.0700 0.073 0.073 0.073 0.073 0.073 0.073 0.0750 0.0750 0.0750 0.0750 0.0750 0.0750 0.0750 0.0750 0.0750 0.0750 0.0750 0.0750 0.0750 0.0750 0.0750000000000

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TABLE X (Continued) 300<sup>0</sup>7 COMPRESSIVE STRENGTH AFTER SINGLE EXPOSURE

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RRECTED KPERTIES (1)	F <sub>cy</sub> psi	26,400 26,100 26,100	24,400 24,400 24,700	20,000 19,300 20,300 19,500 20,300	19,200 19,500 27,200 20,100 19,500
	Fcy psi	26,300 26,100 26,100	22,500	20,300 19,500 20,300 20,300 20,300	19,200 20,200 20,100 20,100
Parts Altra	Actual Test Tep.	301 301 301	30 30 30	<u>88686</u>	<u>888888</u>
	Apeciaen Croup No.	λ Â	\$ <b>3</b> \$ <b>3</b>	112222	333888
ACTUAL REPOSURE CONTITIONS	fotal Strair	600	0.0 235 235	00000	6.3 2 1 1 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2
	a a	333	299	33333	99 <b>9999</b>
	į r	333	8 <b>9</b> 9	55353	888585
	Specimen B.			887 887 87 87 87 87 87 87 87 87 87 87 87	
	Berel Sereis	000	888	00000	888888
NOT LIGNOD	Stree (	000	11.0	30000	
IN SOLE	17-104 1 17-104 1	<b>X</b> .91 <b>X</b> .91 Ję.	N.91 16. X	17.22 17.22 17.24	17. <b>2</b> 17.2 17.2 17.2 27.2 27.2 27.2 27.2 27.2
TRADUC		200	222	22222	<b>8</b> 88888
	į r	સંસંસં	***	zzzzz	××××××
C TH	1 56-51	85 PC 1	II		54

400°F COMPRESSIVE STRENGTE AFTER SINGLE EXPOSURE TABLE XI

<b>•</b>	1 2	**	XXX	×××	488	<b>8</b> 88	877 8	88
TADD		<u>888</u>	222	222	222	888	***	99
20020	117-10-5	999 1997	***	***	15.25 15.25 15.62	17 17 17 17	16. <b>2</b>	17.26
	t) kai	000	0 U <b>0</b>		000	499 888	000	00
2	Totel Strein S	000	000	888	000	888	000	00
	Specimen No.	C3347 C3348 C3348	2112 2112 2112		01154 01154 01154	87113 17113	C534.7 C534.8 C534.8	11135 11135
ACTUAL 1	Tenp.	2272 2272 227	222	<u>888</u>	888 898	388	ភ្មន្ម <u>ថ្ម</u>	82
DOSUER O		888	222	222	223	999	999	23
011101S	Totel Strein S	000	000	0.73 0.95 1.04	000	0.3 0.8 1.0	000	00
	Specimen Group Ko.	888	<b>8</b> 88	22.22	50 50 50 50 50 50	50 80 80 50 50 50	ູລີ	ສສ
PROPERTI	Actual Test Temp.	883	398 7-03 7-03	339 339	286 738 750	888 888	885 777	88
ES AFTER E	F <sub>cy</sub> psi	46,500 47,500 48,100	009°07 009°07 01°000	38,200 36,700 36,600	30,000 30,300 29,100	28,100 27,700 26,300	888 111	18,100
TPOSIRE	Strain Rate in/in/min	0.0085 0.0090 0.0085	c.0065 0.0085 0.0083	0-0075 0-0085 0-0085	0.0067 0.0092 0.0085	0*0080 0*0080 0*0080	0.0075 0.0078 0.0078	0.00 <del>8</del> 5 0.0095
DORRECTE PROPERTUE	Fcy psi	001,811 072,570 112,500	87.57 87.57 88.97	8,1,8 8,7,8 8,00,8	00,00 20,00 20,00	28,100 27,600 26,200	21,300 21,300 21,700	18,100 17,400
S <sup>(1</sup> ) STREET	ar Ta	0.525 0.547 0.547	0.55 0.55 0.55		C.106 0.406 0.392		0.267 0.273 0.292	77.0
TH RATIC	3 <b>9</b> 3:			0.925 0.382 0.621		0.92 0.929 0.882		

(1) Properties corrected to maximal exposure conditions and mondual test temporature.

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where temperatures ( $\pm$  to free societ) were used to compute  $\Theta_{17}$ (I) **Man** 

(2) Conversed to needend test temperatures.

9-5 ъ.° 20.0 0.00 119 0.00000 80.39 88 88 888 89988888 89888888 \$\$,20 80 80 80 88 \$ 888 888 44986655 899888655 exposure canditions. and of each step, and 628 666558 RR RRR 22 20.0 000 , 116-5 EEE. Berthem isolaris strains during each stop. These are listed below nominal (faires ferired by mokinetism of surihitive strain readings obtained at the outy associates.) ลิสิ สสส 333 333988 ខ្លីខ្ល 2288653 200 33 00 000 222 522 522 388 777 \*\*\*\*\*\* 88 57 33 11 10.1 82.24 #25.4 8 0 8°5 8'8 \*\*\*\*\*\*\* 00 a 0 A NA H E and I 23 10 I \*\*\*\*\*\* 00 000 1000 H 39 33 , § \*\*\*\*\* 00 000 , Ì 338 255 33 22 • ĩ 7

ROOM TEMP. TENSILE STRENGTH AFTER SEQUENTIAL EXPOSURES TABLE YI

TABLE XII (Continued) ROOM TEMP. TENSILE STRENGTA AFTER SEQUENTIAL EXPOSURES

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	u ,				145 - 13 44 24 - 23 4 24 - 24 -									
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	1						Co7.0		,	597 597			253 563 769	
र्षे संस	, ) , ) , ) , , ) , , ) , ) , )			30.02	37 38 38 38 38 38 38 38 38 38 38 38 38 38		8. 93 93 93 93 93 93 93 93 93 93 93 93 93	8,100 51,100 52,100		307 <b>1</b> 00	888 1111		200 200 200 200 200	227°55 207°55 207°55
DEAEU PEOPERTI		<b>F5</b> :		306 65	56. 200 200 200 200 200 200 200 200 200 200		8.8	2775 2778		202 11 202 11	38,400 38,200 38,500		121	39,800 39,800
	elor <i>i</i> . 2			11.6	11.5		14-10 11-10 11-10	10.5 9.5 10.5		10.6	10.0 10.0		5.5 9.5 10.0	10.5 10.5
		ps.		0,200	900 200 200 200 200		S. 38	86,100 57,100 57,900		21,100	200 27,200 27,200		X 2000	897.55
	, , , , 10.	ps).		2 009 59	000 000 000 000 000 000 000 000 000 00		890 900	522 522 522 522 522 522 522 522 522 522		88	38,400 38,400 38,200 38,400		121	39,800 39,800
SS IE	: 1000	ъF		e	ក្ខភ្		86	909		22	ជជន		유민문	.! <b>&amp;</b> &
rian tan	-1 8 :1 8	e. E		<b>9</b>  :		,	58°.5			5. 5. 5.			62.5 65.0	\$7.5
* 55135	Rockwe Hartre	<u>.</u> ч		115.0			112-C 115-C			112.C 113.5			ш2.5 Ш.0	ш.5
PRUF	pectar: rout fo			130	200		118			125 126	***		333	333
	Total Strain	×		U	C.15 C.15 C.15		0 <b>0</b>	00 00		00	0.55 0.55		000	5770 5770 9770
	Total 0, -	ູ່ອ		3445-	25.27		15-03 15-03	5.65 5.65 5.65		15-25 15-25	15-25 15-25 15-24		15.21 15.21 15.21	222 282
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TABLE XII (Continued) ROOM TEMP. TENSILE ÉTRENGTH AFTER SEQUENTIAL EXPOSURES

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300°F TENSILE STRENGTH AFTER SEQUENTIAL EXPOSURE TABLE XIII

(Falses derived by settraction of comulative strain readings obtained at the end entry approximate.) (1) Asturi exposure temperatures (2 107 free mechani) ware used to compute 017-

(2) Corrected to nomional test temporatures.

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2	a Specimen	à					221 221	5168 5168 7168		33	iii		
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			100 m	800	8-0 8-0 21-0	1 1 00 1 1 1 1 1 00 1 1 1 1 00 1 1	000	0.15 0.15 0.10	5 18 I	00	6-28 6-28 6-13	1 10 M	00
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۵		3 Step 4							40% 13,500	00	51-0 51-0 51-0 51-0 51-0 51-0 51-0 51-0	12.00	
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TER SEQU	TA SALAR	Accue Test Tes	\$	300	303 303 303	58	88	8 <b>8</b> 8	8	8	8	8	Ŕ
ENTIAL E	IR EPOSIE	l FLY	1 T T T T T T	34 <b>,</b> 900 34,600	32, 700 32, 200 32, 100	305°EE	36,900	37,300	35, 300	34, <b>800</b> 3	2009, 15 2009, 15 2000, 15 2009, 15 2000, 15 200	32,500	34.200 35
XPOSUR	لم ل	Ftu	isq	36,700 36,700	36,600 34,000 30,000	35,500	38,200 38,200	38,200 36,500	37,500	, 100	2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2	8	995 1997
[*.		elong. ir 2in.	-	20.5	20.5 20.0 18.5	18.5	17.5	13.5 12.0		0.22	18.0	84-5 84-5	u c
	OCRAECTES	FLY No. Kinkl	P3: 25:	34, <b>//Ю</b> 35,600 0,520 34,600 35,700 0,520	32,200 34,600 0.522 32,200 33,900	31,500 JS, 700	34,906 36,200 0,5.0 20100 38,300 0,5.0	37,300 36,200 0,557 34,700 36,300	005 12 006 5E	ວິນເຊິ່ງ ແມ່ນ ແລະ ເ	34,600 35,800 0.516 0 31,800 33,300	31,600 33,200 32,500 34,100	
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TR 56		<b>Dini</b>	so <b>di b</b> ij	VILLET CONDITI	ON SHOL				12 4 OF 4	RTIES AFTER I	<b>JCPOSURE</b>			CURIEUTED PROPERTIES (	2) STHEN	JTH RATIC	s.
18 5-585 P	i.	1 1 2 2	t det	Step 2	Stap 3	Step -	Total 01. Encoura	7 Total Strain	Speciaen Group Ko.	Actual Test Temp. OF	r. Pet	Fti: pai	elong. ir 2 ir- 3	Fty Ftu Fsi psi	Hr Ult.	ส้อเรีย รูป ม	10
I t II	PIR		29,500 29,500 29,500 29,500	10% 10% 10%	19,000 19,000 19,000	350 <sup>0</sup> F 24. hr 0 Psú											
		TX8	8	8	8	8	15-38	Ċ	671	30	33,900	35,100	17.5	33,900 35,20	C 0*500 0*70	ē.	
			0-23 25-0 25-0	0.10 0.17 0.18	0.25 0.25 0.11	0 <b>0</b> 0	15-40 15-39 15-35	057 0.68 0.56	149 149 149	858	32,400 31,900 33,400	33 <b>,8</b> 00 33 <b>,2</b> 00	17.0 22.0 18.5	32,400 33,80 31,900 33,30 33,400 34,50	200	0.850 1.850 1.850	ಕಳ್ ವಿಶಿಷ
I	A	-	350 <mark>9</mark> 70 05 71 11	375 <sup>0</sup> 7 3.0 kr 25.000													
63			οu	ია			15.27	00	31 33	ğ ğ	37,500	38,800) 38,600	18.0 19.5	37,600 38,90 37,200 38,70	10 C.561 0.51	4.0	
		02 22 22 22 22 22 22 22 22 22 22 22 22 2	500	1•55 2•2: 1•53			15.27 15.27	1.55 2.26 1.53	ន្តន្តន	ឪឪឪ	36,730	37,300	0 4 0 0 4 0	36, 800 37,40 36,400 37,40 36,700 37,40	999	0.971 0.971 0.976	886 431

TABLE XIII (Continued) 300°F TENSILE STRENGTH AFTER SEQUENTIAL EXPOSURE

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TABLE XIII (Continued) 2007 TENSILE STRENGTH AFTER SEQUENTIAL EXPOSURE

				TADIO I			A STOLT			•					PROFER		Synta Ha	URE	CEREC SACES	111 (2) 21155	STEE	JTH RATIOS
		3	~ 	Stap 3	2 8 8	Step 5	9 6 5	de X	7 Step 8	Step 9	300 10	Total 01. Egosure Elori (1)	Strain Strain	Spealmer Group Fo.	Actual Test Test	F.T psi	າງ ເ ພິ່ງ ພິ່ງ ອ	6.101 %	1		ad ad 4-	
	****	5 107		E S S	884 884 897 897 897	350 2 10 2 10 2 10	325 <sup>0</sup> F 36 hr 19.0	375 <sup>0</sup> 7 3 ht 0 ktá	200 201 201 201 201 201 201 201 201 201	325 <sup>0</sup> 7 36.hr 16.0	375°F 3 Er 0 Esi		1									
an ini	00 ££		80	¥.,	80	80	<b>Б</b> о	80	80	80	50	15-39 15-39	ωu	152 252	ខ្លួន	3 <b>.</b> , 50	35,900 36,60	25.5 28.5	32,25	100 100		
<b></b>	666 ERE	สุสุล	000	899	0.15 0.15 0.10	01.0	88 88	000	<b>1</b> 63	833 823	000	15-39 25.35 36.61	2.6 0.75 0.75	3333	305 105 105	22,000 22,000 22,720	89,65 89,65	14-5 15-5 5-5	31,700 32,000 32,000	200 200 200 200 200 200 200 200 200 200		

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No.         No. <th></th> <th>ž</th> <th></th> <th>THE SHALL</th> <th>NITIONS AND</th> <th></th> <th></th> <th></th> <th><b>BLOK</b></th> <th></th> <th>EPOSUME</th> <th></th> <th></th> <th></th> <th>CORRE PROPERT</th> <th>(5) 18 (5)</th> <th>51</th> <th>RENGTH</th> <th>RATIOS</th> <th>ļ</th>		ž		THE SHALL	NITIONS AND				<b>BLOK</b>		EPOSUME				CORRE PROPERT	(5) 18 (5)	51	RENGTH	RATIOS	ļ
Line         Normalize         Normalis         Normalize         Norm	3	No.		<b>3</b> ep 2	C de X	1 de 8	Tan 9	Lafe *	Spectamo Group No.	Į.	Pet Pet	7tu Pet	elong. in 2 in.	Strain Eato In/in/win	ty ta	rtu ped	Tield	nı.	Rc Yield	ult.
10.0         31.00	1		10.55 10.55	2833 2833																
1         1		ALOIA.	Б	8	8		74-57	0	â	<b>9</b>	38,600	38,600	7.0	0.0078	38,600	38,500	0.574	55 <b>7</b> °0		
1         1		1000 1000 1000	46 000	222 222 222 222	0.70 0.75 0.35	Ì	22.22 22.22 22.22	1.05 0.60	<b>គ</b> គ	338	35.56 36,980 37,180	35,100 36,100 37,100	10.0 2.0 2.5	c0065 0.0065 0.0067	35,500 35,900	35,600 36,000 37,000			0.919 0.930 0.953	0.952 0.952 0.955
1000         150         0.000         39,100         0.000         39,100         0.000         <	3			5 10 R87																
Link       Link <thlink< th="">       Link       Link</thlink<>		9€6₹	o	0	0		24-55	G	Ŕ	1.6E	38,400	40,000	7.5	0°.0093	38,100	39,700	0.563	0.51		
1.13       1.000       1.100       1.100       1.100       1.100       0.100         27:5       2.10       2.10       2.10       1.100       1.100       1.100       0.100         27:5       2.10       1.100       1.100       1.100       0.100       1.100       0.100         21:33       0       0       0       1.77       4.02       33,500       3,500       3,500       0.103         11:100       0       0       1.77       4.02       33,500       31,500       0.133       0.133       0.133         11:100       11:100       31,500       1.17       4.00       1.17       4.00       0.133       0.143       0.143       0.143       0.143       0.143       0.143       0.143       0.143       0.143       0.144       0.144       0.144       0.144       0.144       0.144       0.144       0.144       0.144       0.144       0.144       0.144       0.144       0.144		133	01.0 01.0	0-15 0-15 0-15	8 500		<b>388</b>	0.25	នំនំនំ	<b>35</b>	36,400	36,400	 	0.0090 0.0092 0.0090	37, 100 36, 400 35, 400	34,500 37,500 36,300			6.974 0.955 C.926	0.972 0.945 0.945
1273         0         C         14.%         0         127         4.02         33,500         34,100         0.4480         0.433           41233         0         C         14.%         0         127         4.02         33,500         34,100         0.4480         0.433           41231         0         C         14.%         0         127         4.02         33,500         34,300         34,300         0.4430         0.433           41231         0.25         0.10         14.96         0.35         127         4.01         31,500         31,900         34,300         0.4430         0.435           41231         0.25         0.2004         31,900         31,900         32,900         14,900         32,900         0.442         0.435           41233         0.25         0.2004         31,900         32,900         12.0         0.994         0.947         0.445         0.947           41233         0.210         14.00         31,700         32,700         12.0         0.994         0.994         0.944         0.944         0.944           41233         0.210         127         4.01         31,700         32,700         12.0	3		88%I	5 10 1 10 1 10																
<b>11.201 0.25 0.10</b> <b>11.900 31,900 31,900 31,900 32,400</b> U. <sup>45</sup> 2 0.947 <b>12.21 0.25 0.10</b> <b>14.96 0.30 127 4.02 31,700 32,200 12.0 0.0097 31,800 32,300 0.949 0.949 0.949 0.949 0.944</b> <b>12.239 0.200 0.10</b> <b>14.96 0.39 12.7 4.00 31,700 32,200 12.0 0.0094 31,900 32,300 0.949 0.944</b>			• •	00			77 8.8 8.8	<b>0</b> 0	55	<b>3</b> 3	33 <b>,2</b> 00	75.00 77,000	н 19 19	0.00 <del>9</del> 3 0.0096	33,300 33,600	34,100 34,300	0°1780 0°1780	0.433		
		a da	200 200				5 <b>8</b> 8 777	0.30 0.30 0.30	555	<b>4</b> 84	500 200 200 200 200 200 200 200 200 200	2000 2000 2000 2000		0.0096 0.0097 0.0094	31,900 31,900	32,300 32,300			u. 452 0.949 C.945	0.947 0.944 0.944

TABLE XIV 400°F TENSILE STRENGTH AFTER SEQUENTAL EXPOSURES

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Spectmen invlantic strains during each step. These are listed bulles noninal exponence conditions. (Falmes derived by subtraction of commistive strain readings obtained at the end of each step, are only approximate.)

(1) Actual exposure temperatures (  $\pm \, k^0 f$  free meeting) were used to compute  $0_1\gamma \cdot$ 

(2) Corrected to reached test temperatures.

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(2) Corrected to pominal test temperatures.

.**3**.1 1 CORECTED FROFEREES - 2) e në pşi 4°4 4 ä 113/57/44 elong. Strain in 2 in. Raie TABLE XIV (Continued) 400°F TENSILE STRENGTH A FI ER SEQUENTIAL EXPOSURES 1 **7** 2 PIOPERTAS ATTA ELPOSINE 5 Spectamn Croup No. later States [1] 4 4 5 10 4 5

C. - 18 C. - 938 1.070 0.931 0.950 ы. Ч. STREASTH WALLOS 0.932 0.932 0.932 0.932 0.932 1.023 C.946 O.957 919 19 19 19 0.382 0.382 0.372 0.372 0.372 0.376 C.330 0.341 C.352 3 0.430 0.420 0.420 0.420 0.420 323332 388333 323333 333333 88 888 88 888 88 8 8 8 8 C.00% C.00% O.00% O.00% O.00% O.00% ะ รูชาวาร อาราชอารา รูชาวาร การเรา 5.7 .00° 888 888 888 888 888 88 9**88** 88 8**8** \*\*\*\*\* 38 485 <u>NNSSSS SEENNN</u> 3<u>5</u> 333 00000 00000 0.0 4 4 4 X 8 8 \*\*\*\*\* **77 77** 6260 00 MAN NORTH REPORTE CONTINUE AT \*\*\*\*\* co 999 ..... 2.44.1 1.44.1 AND OF SAN RARI co SPS 19999 999886 22 **22** 3 7

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TABLE XV ROOM TEMP. COMPRESSIVE STRENGTH AFTER SEQUENTIAL EXPOSURES

			NONC XAL Z	REPORTE COM	DITIOUS AUC					FROPER	LFS 821	ET LOCUTE	б: 	CORRECTED ROFERTIES (2)	
		i i	845 I	Stap 2	Step 3	Step 4	Total 017 Esposure El0"3 (1)	Strun	Specimen Group Ro.	Rockne Hardne Rp	18 <sup>8</sup>	Actual Test Temp. OF	rey tsi	F cy F s 1	e:
	3		350 <sup>7</sup> 3.5 F	335 <b>9</b> 20 M 33.5 M	27.5 20. kr										
		6227 6723	50	80	<b>8</b> 0		88 77	00	176 176	2.311 2.311	76.5	75 75	t.,700 et,500	6., TCC 86, 500	កាស់ (សំ មួយ។ ព្រំបំ
33:       33:       33:       33:       33:         33:       33:       33:       33:       33:       33:         33:       3		5211 52211 52211	0.25 0.25 72.0	0.35 0.37 0.31	0-35 0-45 0-42		808 RFF	<b>1</b> 88	176 176 176	15.0 15.0	2005 2005	75 25 25	60, 530 58, 600 5-300	60,500 58,600 59,300	· · · · · ·
	7		1.5 M	315 <b>9</b> 2015 2015 2015 2015 2015 2015 2015 2015	2759 100 kr 26.5 kri										
33:5:       33:5:       33:5:       33:5:         1:       33:5:       35:5:       35:5:       35:5:         1:       33:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:         1:       35:5:       35:5:       35:5:       35:5:     <			55500 55500	963 963	1000 1000 1000		38388 66666	388 388	letet	nés nés nis s s s s s s s s s s s s s s s s s			81.255 81.255 81.858 82.858	ស្នានស្នាស ភូមិស្នីស្នីស្នី ភូមិស្នីស្នីស្នី	5 0 8 0 8 0 0 0
1       1	1		100 100 11 11 11 11 11 11 11 11 11 11 11	25.5 21.5 21.5 21.5	400 <sup>7</sup> 7 3.5 hr 19.5 km										
L. 1007 3517 1007 3507 35.5 bit 21.5 bit 25.6 25.5 bit 25.6 25.7 bit	:		<b>5.3</b> % 5.50 5.50	0000 0113 0113	0000 200 200 200		222222 222422	<b>888</b> 8888 8997	33333		61.5 56.5 57.0 57.0	55555 555555	43,800 42,700 40,500 39,900 39,900	12 12 12 12 12 12 12 12 12 12 12 12 12 1	:::: :::::::::::::::::::::::::::::::::
GJ27       5       C       0       15.42       0       182       111.0       59.5       75       42,300       42,300       5.69         GJ37       C       C       0       15.42       0       182       111.0       59.0       75       42,300       5.69         GJ317       C.25       0.11       0.22       0.17       15.40       0.77       182       109.5       55.0       75       37,300       5.573         GJ317       C.25       0.11       0.22       0.17       15.40       0.77       182       109.5       55.0       75       37,300       5.573         GJ317       C.25       0.11       0.22       0.17       15.40       0.77       182       109.5       54.700       37,700       35,700       5.573         GJ317       C.25       0.13       0.22       0.117       15.40       0.77       182       109.5       54.700       36,700       5.577         GJ312       0.13       0.23       0.154       0.17       182       109.5       54.700       36,700       5.677         GJ312       0.13       0.14       182       109.5       54.00       7.73       36,700 <td>4</td> <td></td> <td>300 100 kr</td> <td>351<b>9</b> 24. hr 215 hri</td> <td>19-5 M</td> <td>25.0 F</td> <td></td> <td></td> <td></td> <td></td> <td></td> <td></td> <td></td> <td></td> <td></td>	4		300 100 kr	351 <b>9</b> 24. hr 215 hri	19-5 M	25.0 F									
			56560 1933	00000 1811	k 13 k 0 0 0 0 0	0 0 11 0 11 0 1 1	2122222 2123222 213323	<b>7</b> 77	aaaaa		5.00 5.00 5.00 5.00 5.00	****	12,500 33,700 39,700 39,700	42,500 39,500 38,700 80,600	0-545 0-572

with approximate.) (1) Actual copower temperatures ( $_{\Delta}$  M<sup>of</sup> frem monimal) were used to compute  $\Theta_{17}$ . (2) Corrected to maximal test temperatures.

(2) Corrected to mainal test temperatures.

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TABLE XV (Contrued) MOOM TEMP. COMPRESSVE STRENGTB AFTER SEQUENTAL EXPOSURES

			EPOCHE CON	DITIONS AND					FROFE	LA SIL	THOCHE		CORRECTED PROPERTIES (2)	961U	
Sequence Cede	29. 29. 29. 20. 20. 20. 20. 20. 20. 20. 20. 20. 20		2	С ¥8	Stap 4	Total 017 Exposure 110")	Total Service	Spectamn Group No.	Rockre Hardty	្នុន្ទ្	Actual Test Tent.	E E	Fc7 Fu1	ta ar	۲
4			242 272 2	Local Local Line F	3500 <del>1</del> 24. 14 145 145										
		80	80	50	<b>8</b> .	15.43	<u>ا</u> د د د	ăă.	नुसं		2 2 5	ងដូរ ទទ័ន	ង ខ្លួន ខ្លួន	5.5	ی ن
	533 533	ಸ್ <b>ಕ</b> ತ ಕ್ರಿಕೆಕ	6-0 (1-0	(10 (10 (10 (10 (10))	5 <b>8 8</b> 5 <b>6</b> 6	525 151			ง เมื่อ เปล้ เมื่อ เมื่อ เมื่อ เมื่อ เมื่อ เมื่อ เมื่อ เมื่อ เมื่อ เมื่อ เมื่อ เปล้ เป เป เป เป เป เป เป เป เป เป เป เป เป	C		28 11 11			00
F			2507 215 F	10-5 H	10 10 10 10 10 10 10 10 10 10 10 10 10 1										
	19999	8 <b>6 8</b> 6 6 6 6 6	6660 66600	<b>KR</b> <i>I</i> 00000	<b></b>	1999 1999 1999 1999 1999 1999 1999 199	<b>ាងដ</b> ចំព័ត៌ចំន	FFFFF	110.5 110.0 110.0	88999 8899 8899 8999 8999 8999 8999 89	*****	17 27 28 28 28 28 29 29 29 29 29 29 29 29 29 29 29 29 29	2002227 2012 2022 2022 2022 2022 2022 20	c. 575 c. 572	201
4		L 1 1 ARo	H LAN												
			55000			2.2.2.2.2 2.5.2.2 2.5.2.2	နိုင်္နီ ဦးဦးနိုင် ကို ကို ကို	<b>ពន្ល</b> ន្លន៍ន៍ន៍		52.5 56.5 56.5 56.5 57.5	121212	1,20 1,20 1,30 1,30 1,30 1,30 1,30 1,30 1,30 1,3	222 222 222 222 222 222 222 222 222 22	c. 581 c. 572	

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			2 2		4438 1997	204033 27 - 22 - 2
		12:500	5 8		8398 7777	232388 
		17 HEL			Cuat	Seettt
			73		0; ; 0; ;	
		1014			110.5	134 138 138
	GURES		Speciality		1233	zzzyła
	IT EXPO		Strein			44888 6666666
	RUENTU		Total 9,7		1122	755799
	Almond) L'TTER El		Star 10	19:50	1000	000000
	I TA (COL		Step 9	25.04 26.04 26.04 22.0 Mil	3000	111 937932 000000
	TABLE TABLE		- 4-16	188 188 198 198 198 198 198 198 198 198	6000	61988×
2 A 1	COMPRI		Stop 7	1920		800000
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	DOM	CONTINUE	5 9015	12 22 22 22 22 22 22 22 22 22 22 22 22 2	สียงบ	31.123
		L LUCCIE	Stag =	202 202 202 202 202 202 202 202 202 202	3000	112523
		77(2)05	ŭtep .	275° 275° 22.5 Kel	ວັບບຸມ	,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,
			Juep 2	5 A 12 2 A 12 2 A 12	สียงอ	u <b>-3363</b>
			Blap :	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	ძაად	333333
						K20555
			<b>]</b> ;	I		

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	EXPOSURES
	SEQUENTIAL
TABLE XVI	VE STRENGTB AFTER
	300°F COMPRESS

		SSR I TYNDION	IPOSUME COM	CITA CITATION AND AND AND AND AND AND AND AND AND AN					PHOFEC	- Center Center	ooraected Boffertes (2)	NGELS	JOLINA HI
Sequence	Pociano Ilo.	94 B	Step 2	5 <b>6</b> 3	- 1 de 25	TOLAL 017 Exposite 210	1919 1919 1919	Speciam Group No.	Actual Test Temp.	F cy pat	f cy pei	a.	e.
3		350°F 3.5 hr 31.5 hrt	33.5 H	2024 2025 24.5 H									
	South States	ង១១១១ ភូនខ្លួន	80008 8660	88% 50000		4444 88688	с 0.78 0.78 0.78	801 801 801 801 801 801 801	88888	8,53 2,73 2,73 2,73 2,73 2,73 2,73 2,73 2,7	52,58 53,58 53,58 58,58,58 58,58,58 58,58,58,58 58,58,58,58 58,58,58,58,58,58,58,58,58,58,58,58,58,5	.723	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1
1		350% 3.5 % 27.5 kt	335 <b>9</b> 20 14 26 144	235 200 F 24.5 F									
		1111 555000	6000 50000	1999 1999 1999		***** *****	87.7 000000	<b>6</b> 666666	ខ្មន្តខ្មន្តខ្	8,52,52 8,52,52 8,50,50,50 8,50,50,50 8,50,50,50 8,50,50 8,50,50 8,50,50 8,50,50 8,50,50 8,50,	8,88,88 8,68,88 8,68,88	ELT. 257.	68.5° 68.5° 68.5°
<b>1</b> 1		300 <sup>4</sup> 70.5 10 10 10 10 10 10 10 10 10 10 10 10 10	350 <b>°</b> 22.5 <b>E</b> E	100 <b>7</b> 3.5 km						•	<b>x</b>		
	6314 (5) 63146 (5) 63146 (5) 63147 (5)	38 33 0 0 0 0	ង។ 2000	0.35 0.39		15.22 15.22 15.32	8 R 0 0 0 0	8888	8828	33,500 33,500 33,500	33,500 33,500 33,500 33,500 33,500 33,500	<u>4</u> 8.	563°
1		20.5 H	350 <b>7</b> 25.5 <b>1</b> 21.5 <b>1</b>	100% 3.5 hr 19.5 hr	200 7 200 1 1 200 1 1 200 1 1 200 200 200 200								
	189993 18993	ភូនភូ ទទទទ	<b>FR</b> 2001 - 0	<b>ភ្លេក្</b> ទទទទ	<b>ក</b> ្រុង 200000	4444 4444 4444	<b>ដដែន</b> ចំព័ត៌ចំព		55835	%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%	36,30 38,65 33,65 32,65 32,65 32,65 32,65 32,65 32,65 32,65 32,65 32,65 32,65 32,65 32,65 32,65 32,65 36,30 36,30 36,30 36,30 36,30 36,30 36,30 36,30 36,30 36,30 36,30 36,30 36,30 36,30 36,30 36,30 36,30 36,55 37,655 37,055 37	267°	617 158 158
• Here Here Here Here Here Here Here Her	an inclarit	c strains di subtractic	uring each i ar of cumula	rtop. These stive strain	are listed reatings	below nomin trained at t		e conditions. each step, are					

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Actual exponent temperatures (2 A<sup>op</sup> from nominal) were used to compute 0<sub>17</sub>.
 Corrected to numbral test temperatures.
 Ernds refer specimens (51)7 and 0313H were 0.013 and 0.013 fa/in/mis respectively.
 Ernds were 0.007 - 0.001)

TABLE XVII TENSBON CONTROL TESTS FOR COMPRESSION PROGRAM

	1157月1		•		FUSITE JOYS			347764		III EII CSUI	ų		CORFECT PROPERTY	គ្រង	STREET	EATTOS
<b>i</b> .	<b>A</b> 11	T(170105 t)	Speciaen No.	į	1 h	Spectreen Group No.	Nockwell Mardness R <sub>1</sub> R <sub>2</sub>	Test 1		ty si	Ftu psi	elorg in 2ir. S	FtJ psi	Ftu psi	Yield	l'Itimate
88		12.52 12.52	• E.C.	<b>8</b> .8 ~ ~		33	117.0 80.	0 v	55	8,300 8,900	77, 300 78, 200	9.6 9.5	ж, 30 96, 50	77,300 74,200	C.999	1.Clt 1.C28
**	ន្តដ្ឋ	9EI 9EI	• 3552 2335 •	25 2 <b>5</b>	88	172	117.C 81. 117.0 82.	<i></i> 66	0.0	2,200	25 <b>,9</b> 00	0.11.0	of,500 o7,100	75 <b>,</b> 500 76,500	c.987 c.981	566°0 166°0
XX	ន្តន្ត	14	4 1220 4 1220	ថ្លត្ថ	88	61 13	116.0 75 115.0 77	~~~ ~~~	v v	1,500 0,700	71,100	9•5 5•5	51,800 21,000	2011 L	C.90L 0.892	6.538 C.538
žž	99	8.5. 7.7	12115 CLISE •	хx	33	E E	106.0 78 106.0 78	00	~~~ ~~~	5, 500 7, 200	70, 50C	10.5	57 <b>,</b> 500	71, COC 55, 166	C.865 C.845	. c.935
<u>88</u>	ន្តន	***	C7267 77265	хă	9 9 9	ลัล		 	t t. د ب	9°70 200	50,300 49,800	14°C	30°°35	30 <b>.</b> 900	0.7J 12.2	C.665 0.458
eee	~~~	997 1977 1977	C6.201 + C5.201 + C6.208 +	E E E	<b>ᲝᲝ</b> Ო	ន័និន័			000	92.4 92.4 7	65, 800 500, 63 500, 63	ू २००	55,300 56,000 57,700	5,730 6,730 9,700	c.810 c.350 c.860	C.88C C.92C C.91C
ХX	88	15.21 15.21	3125 9126	ы К	84	LES LES	112.0 61 112.0 61	ن.ن م بي	1 -1  	0,100 0,100	20: 55 20: 55	5-5 10-2	201 <b>0</b> 201 <b>0</b>	55,3CC	0-59 C-58	10.0
នុដ្	22	15-48 15-48	• 8577 1728 •	8 <u>8</u>	20	178 178	ш.º 59 Ш.5 60	00	5 10 10 10 10 10 10 10 10 10 10 10 10 10	9,100 9,400	77 200 200	5.5 9.5	39,100 39,400	2000	0.572	(.115 0.121
88	22	17.25	C6157 • C6158 •	ĕĕ	22	17 17	28.2 5.88 5.88	ó.i L.	<u>14 14</u>	9,700 9,300	35,300 36,600		19,700	36,300 300,96	0.288 0.282	0.516 C.520
88	27	17.26	C7273	ãã	33	<b>X</b> X	1 1 1 1 1 1 1 1	**	04	8,500 8,200	25,500 25,200	35.0 35.0	18,500 18,200	25.30 25.30	0.270 C.260	0.335 0.334
gy	22	17. <b>26</b> 17. <b>26</b>	CT2LT CT2LE	ãã	22	ጞጞ	1 8 1 1 1 1 1 1	33	99	,6,200(1)	17,200	47.0 48.0	16,200 16,400	17,100	0.237	6.225 C.226

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Specimere pulled with flat grips.
 Strait rate = 0.012 in/in/min

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## **ORIGINAL STRENGTH AND STRENGTH AFTER REFERENCE EXPOSURE** TABLE XVIII

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		OFFICE	AL STRENGTH	1	STR REFER	ENGTH AFTE	<b>д</b> URE (2)	CHAN	IGE IN STREW	ICTTH
		μ, Γ	r (1)		)	(1)		Ff -	Fl	
	Sheet	Tensile Yieli	Tensile Atimate	Comp. Tieïd	Tensile Yisld	Tensile Ultimate	Comp. Tield	Tensile Tield	Tensile Ultimate	Comp. Yield
8.		pei	psi	<b>te</b> i	psi	jsi	isq	isq	1sd	j <b>s</b> d
E.T.	<b>⊲</b> βΟ	68,300 67,000 68,400	78,300 76,400 76,100	- 74,,300	25,800 26,000	43,600	- - 28,300	42,500 41,000	34,700 32,000	- 000, 64
8	<b>≺ A</b> U	65 <b>,</b> 000 -	- - -	_ _ 70,700	24,500 - -	38 <b>,0</b> 00 - -	- - -	- - -	31,900 - -	- - 43_300
8	<b>≺</b> Ø U	57,000 54,400 56,400	58,300 56,200 58,100	- - 63,100	24,400	28,400 28,500 -	- - 300	32,500 30,000 -	29,900 27,720	- - 36,800
8	4 19 0	42,000 42,700 43,100	413,600 44,200 44,400	- - 47,500	20,300 20,700	20,400	- - 21,600	22,000	23,200 23,011	

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(1) Average Strength Values. Refer to Tables III through XI

(2) Reference Exposure =  $450^{\circ}$ F - 10 hr.

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(i) (i) STRENGTH DETERIORATION FACTOR FOR STRESSED EDFOSUAL 1.05 INLASTIC STEALS 1. 1. j. 9 9 9 9 0 0 0 0 53.1 2.2.1 2.2.1 1 1 1 1 1 1 C.255 0.123 6.5 3333 \*\*\*\*\* .... Fer = F. : . • Tersile Lituate 1.036 6.63 0.712 0.712 125-0 C.5% 1.21 6; ; ; ۰. الله 0.2194 99995 9995 -0.165 .... Tersile Yield 55.1 C.98. 0.952 0.976 20.00 0.658 - <del>R</del> - <del>R</del> - 0 - 1 267-0 \*\*\*\* 6663 4999 6.17 10.1 ; : Ì . STRENGTH DETERIORATION FACTOR FOR STRENGED ELPOSURE 0.25 INCLASTIC STRAIN See See C.985 1.005 0.915 0.978 118-0 181.0 1 1 + 5.50 0.230 3353 9999 8888 1 1.1 . D - Fet - F1 Tersile Ultimate 0.968 0.969 0.982 0.753 0.712 0.760 0.761 0.63.0 1 0.510 0.339 55.50 55.50 56.50 0.109 1 1 1 1 1 1 50-T 0-669 6688 99999 -0.155 1 i . Tensile Yield 88 1.000 8.° 8.8 1 0.765 0.795 0.795 0.795 0.712 125.0 88.45 99999 99999 0.160 181 . 11 tor P1 and Pr. Pj. Multiplied by appropriate Re from Pigure 16. Tempile data are from Shoot A tests only. STRENGTH DEFICIENTION FACTOR 986.09 98.00 91.00 91.00 Tield. 0.9%6 1.52 0.8.3 1 1 1 1 1 1 0.352 1 61.0 353 . . . 3 0000 3 0 - <u>7 - 7</u> 7 - 77 Tensile Ultimate 0-337 0-262 0-250 236 ÷ 0.968 0.615 0.771 0.612 0.6312 0.678 18.9 0.729 9.X.0 2.X.0 E al 11.0 . 0000 Tread. 0.80 0.972 0.00.00 62.0 c.555 0.606 2.0 35£3 0.752 ..... 0.155 12.0 0000 3,70 283 283 67,100 191.2 3833 38.38 8888 8888 8888 8888 j \* \* \* 1 Ĩ STRATCHAR TROPERATURE Arries stands whus - refer to Tables IT through II. Mer to estima 5 through 7 for Per and to Table Mill 1 Per for stressed supreme is Per visitoreed supreme Tradio Ditiente 88 58 58 888 888 888 5388 8888 66,90 5258 5258 5258 8.1 8 8 8 8 8 88 5.20 888 5×1 8888 7788 8: 34 8 8 3 ž Teles a 3 X 1 8 8 8 888 8 227 2 55,200 19,200 60,109 325 288 57,800 191 88 98 98 89'3' 89'3' 8888 8888 8888 8888 20, 21 . ī İs 18 188 18 1888 1888 1888 1888 1888 18 5 223 HINDER U 8.27 5.5 67-53 3333 \*\*\*\* 7777 \*\*\*\* \*\*\*\* 2.2 ADDRIAL CONTINUES Ĩ ł 53 ļ 888 8888 2222 2222 88 2222 2222 a a 1 5 88 222 8888 2222 88 88 8888 3333 8888 **3333** 222

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TABLE XIX STRENGTH DETERIORATION FACTOR FOR SINGLE EXPOSURE - UNSTRESSED AND STRESSED

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TABLE XX STRENGTH DETERIORATION FACTOR FOR MULTIPLE EXPOSURES - UNSTRESSED

1.

$F_{eff}$ $I_{eff}$ <				AFTER EXPOS	EMPERATURE URE		STRENGTH L	ETERIORATION	FACTO SURE
Sequence         Total $\frac{1}{2}$ , Test         Tensile         Tensile				Fef	(2)		Da Fer F	$\frac{-F_1}{-F_1}$ (3)	
A-1 $11,000$ $11,700$ $65,600$ $0.828$ $11,1,600$ $1000$ $50,500$ $50,500$ $50,600$ $50,600$ $0.828$ $11,1,600$ $100$ $33,600$ $50,500$ $51,900$ $51,900$ $0.828$ $11,1,600$ $100$ $33,600$ $33,600$ $33,600$ $51,900$ $0.828$ $11,1,600$ $100$ $38,100$ $39,700$ $71,500$ $66,000$ $0.823$ $11,1,600$ $10,00$ $38,100$ $39,700$ $71,500$ $66,000$ $0.823$ $11,1,600$ $10,-7,500$ $59,1000$ $39,700$ $70,300$ $-0.745$ $11,-600$ $10,-7,500$ $59,1000$ $70,300$ $-0.245$ $0.2803$ $11,-200$ $11,-600$ $10,-7,100$ $59,1000$ $0.2800$ $0.2803$ $11,-200$ $11,-200$ $10,-7,100$ $10,-7,100$ $0.2800$ $0.2800$ $15,-300$ $10,-7,100$ $11,-200$ $11,-200$ $0.2800$	Sequence	Total 917 11073 917 (1)	Test Temp. OF	Tensile Tield psi	Tensile Ultimate Psi	Comp. Yield psi	Tensile Yield	Tensile Ultimate	Comp Yiel
$A^{-2}$ 14.60 $B^{-1}$ $14.60$ $B^{-1}$ $59,900$ $71,500$ $55,600$ $52,400$ $53,500$ $0.803$ $114.60$ $100$ $38,100$ $39,700$ $53,500$ $0.820$ $0.820$ $114.60$ $B.$ $T.$ $57,500$ $69,100$ $ 0.820$ $N^{-1}$ $11.60$ $B.$ $T.$ $57,500$ $69,100$ $ 0.820$ $N^{-2}$ $11.60$ $B.$ $T.$ $59,900$ $70,300$ $ 0.745$ $N^{-2}$ $114.60$ $B.$ $T.$ $59,900$ $70,300$ $ 0.745$ $N^{-2}$ $114.60$ $B.$ $T.$ $59,900$ $70,300$ $ 0.202$ $N^{-2}$ $114.60$ $B.$ $T.$ $43,300$ $0.203$ $0.200$ $N^{-1}$ $115.30$ $300$ $B.$ $0.1400$ $ 0.200$ $0.202$ $N^{-1}$ $15.30$ $0.00$ $0.0100$ $0.0100$ $0.0250$ $0.0149$ $N^{-1}$ $15.30$	Į-I	09•17 09•17 09•17	в. Т. 300 400	61,000 50,500 33,600	71,700 50,600 38,600	65,600 53,900	0.828 0.800 0.843	0.809 0.742 0.784	18.0 - 75
A'-1       1460       R. 7.       57,500       69,100       -       0745         A'-2       1460       R. 7.       59,900       70,300       -       0602         B-12       15.00       R. 7.       59,900       70,300       -       0602         B-12       15.00       R. 7.       59,900       70,300       -       0602         B-12       15.00       R. 7.       445,800       60,400       -       0495         B-12       15.00       300       33,500       34,3200       -       0495         15.00       300       34,700       34,3200       -       0508       0596         15.30       8 T.       41,400       56,800       43,3700       0343       0508         15.30       300       34,700       36,600       37,800       0343       0495         15.30       15.30       8.0       43,900       56,800       43,370       0343         15.30       15.30       8.0       56,800       43,370       0343       0495         15.30       26,900       36,600       28,900       28,900       0343       0495         15.30	A-2	99•77 99•77 99•77	R. T. 300 400	59,900 50,600 38,100	71 <b>,</b> 500 52,400 39,700	<b>66</b> ,000 53,500	C.802 0.803 0.820	0,804 0,802 0,831	0.81 0.73
A <sup>1-2</sup> $1460$ $B_{\bullet}$ $7, 39, 900$ $70, 300$ $-10, 200$	1-1A	09ייד		57,500	001.69	ı	0•745	0.734	I
$B-12$ 15.00 $B_{-1}$ $46,800$ $60,400$ $-10,495$ 15.00       300 $42,400$ $43,200$ $-0.495$ 15.00       300 $42,400$ $43,200$ $-0.495$ 15.00 $15.00$ $400$ $33,500$ $34,200$ $-0.608$ $B^{-1}3$ $15.30$ $200$ $34,700$ $56,800$ $43,300$ $0.376$ $15.30$ $300$ $34,700$ $36,600$ $43,300$ $0.343$ $0.608$ $15.30$ $300$ $34,700$ $26,600$ $43,300$ $0.343$ $0.608$ $15.30$ $300$ $34,700$ $28,900$ $28,900$ $23,200$ $0.343$ $15.30$ $15.30$ $100$ $38,200$ $-13,130$ $0.437$ $15.30$ $300$ $37,100$ $38,200$ $-13,130$ $0.437$	A*-2	09•17	њ. -	59,900	20,300	ł	0.802	0.769	I
B-13       15.30       R. T.       41,400       56,800       43,300       0.376       (         15.30       300       34,700       36,600       37,800       0.343       (         15.30       400       28,900       29,400       -       -       0.343       (         115.30       400       28,900       29,400       -       0.437       (         115.30       8.0       37,800       38,200       -       0.437       (	<b>B-1</b> 2	15.00 15.00 15.00	в. Т. 300 4,000	46 <b>,800</b> 42,400 33,500	60,400 43,200 34,200		0.495 0.550 0.608	0•484 0•494 0•594	111
B <sup>1</sup> -13 15.30 B. T. 43,900 58,200 - 0.437 ( 15.30 300 37,100 38,200 - 0.437 (	<b>6-1</b> 3	15.30 15.30 15.30	в. Т. 300 400	41,400 34,700 28,900	56,800 36,600 29,400	43,300 37,800	0.376 0.343	0.388 0.292 -	0.326
	B'-13	15.30 15.30	в. <b>т.</b> 300	43,900 37,100	58,200 38,200		0.437 0.423	0•431. 0•350	11

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Average values of  $\Theta$ , carried to 3 significant figures only. Averages of individual strength values from Tables XII through XVI. Tensile specimens for A sequences and B-12 sequence were taken frum sheet A. Rest of tensile specimens (above) were from Sheet B. Refer to Table XVIII for values of  $F_1$  and  $F_1$ . Θ

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 TABLE XX (Continued)

 STRENGTH DETERIORATION FACTOR FOR MULTIPLE EXPOSURES - UNSTRESSED

			STRENGTH A	T TEMPERATURI ECPOSURE		STRENGTH DH FOR UNSI	ETERIORATION RESSED EXPOS	FAC TOR URE
			ſ4,	af (2)			f - F ((	3)
Sequence	Total <b>9</b> 17 200-3 (1)	Test Temp. op	Tensile Yield Psi	<b>Tensile</b> Ultimate psi	Comp. Yield Psi	Tensile Yield	Tensile Ultimate	Comp. Yield
Ъ-ц	15.40 15.40 15.40	R. T. 300 400	37,900 34,700 25,700	53,400 36,000 26,100	42,400 36,500	0.290 0.343 0.245	0.281 0.270 0.213	0.305 0.277
B-24	15.40 15.40	R. T. 300	37,900 34,100	53 <b>,</b> 000 35 <b>,</b> 500	42,800	0.290 0.323	0.265 0.253	0.315
B-13a	15.30	R. T.	001,14	56,200		0.368	0.369	J
B-14	15.40 15.40	R. T. 300	39,600 33,900	54 <b>,</b> 300 35 <b>,</b> 100	42,500	0.332 0.317	0.309 0.238	0.308
A	15.30 15.30	R. T. 300	44,200 37,500	58,800 38,800	42,900	0.437 0.437	0.450	c.317
<b>B-1</b>	15.40 15.40	R. T. 300	38,900 34,300	54,300 36,100	42,400	0.315 0.330	0.309 0.274	0.306 -
3-2	15.40	R. T.	39,900	24,600	ł	0.339	0.319	,
<b>8-</b> 3	15.40	R. T.	38,700	53,900	I	0.310	0• 25 /	ı

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CONFRESSION SPECTMEN

FIGURE 2 TEST SPECIMEN DETAILS

TICKILE SPECINEN



**BUDS:** Flatness <u>+</u>.0005 Squareness <u>+</u>.0005 Parallel <u>+</u>.0005 бо. +

SIDIS: Parallel



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## FIGURE 3 TENSION SPECIMEN EXPOSURE ASSEMBLY

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цt	78.5 78.4		78.5 78.8		78.1	70.8 7.77	
,,,,	<del>7</del> 77					24	
r.	69°.0		1-69 1-69		68-7		
nt.		1-17 1-17		76.7 78.1		78.9 79.3	si
							д 100
PL		67.7 67.1		65.8 67.5		0-89 4-89	<sup>F</sup> t1 80,500 76,700
đ	79.1 79.9		78.9 78.9		77.3 78.1		pa1
111							\$88
PLI	68-8 69-2		68.3 69.4		67-5 67-9		70,8 65,3
1T		77-8 78-5		7.0		0.5 79.3	at mum mutan
-							문문
							<b>K</b> A
74		\$ \$ \$		66.1 65.3		0.6 2	Sheet (Above
UIt	78.3		76.6 76.8		7.5		
H							
PCL	<b>68.0</b> 67.5		68.2 68.2		67.6 67.7		
Шţ	78.5 78.0			78-1 77-5		67.5 77.5	
						四권권	
2	68.7 67.9			6e.0 67.5		67.2 78.5	

FIGURE 11 TENSILE STRENGTH VARIATION THROUGHOUT ONE SHEET OF TEST MATERIAL

76, 700 78, 300 77, 000

(Above) Hinimum 65,300 Average 68,300 Approx. universal average 67,000 for 0.063 in.7075-T6 Alclad

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## FIGURE 12 COMPARISON OF STRENGTH AFTER UNSTRESSED EXPOSURE FOR TWO LOTS OF 7075-T6

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FIGURE 13 CORRELATION OF STRENGTH REDUCTION AFTER EXPOSURE WITH EXPOSURE PARAMETER  $\theta_{17}$ 

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FIGURE 14 (Cont.) EFFECT OF EXPOSURE-ACCUMULATED TENSILE STRAIN ON SUBSEQUENT STRENGTH ---SUNGLE EXPOSURE

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FIGURE M (COML.) EFFECT OF EXPOSURE-ACCUMULATED TENSILE STRAIN ON SUBSEQUENT STRENGTH ... SUGLE EXPOSURE

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FIGURE 15 (Cont.) EFFECT OF EXPOSURE-ACCUMULATED TENSILE STRAIN ON SUBSEQUENT STRENGTH ---MULTIPLE EXPOSURE



FIGURE 15 (Cont.) EFFECT OF EXPOSURE-ACCUMULATED TENSILE STRAIN ON SUBSEQUENT STRENGTR ---MULTIPLE EXPOSURE

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FIGURE 17 (Cont.) STRENGTH REDUCTION AFTER UNSTRESSED AND STRESSED SINGLE EXPOSURE

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FIGURE 17 ( Cont.) STRENGTH REDUCTION AFTER UNSTRESSED AND STRESSED SINGLE EXPOSURE



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## COMPARISON OF STRENGTH AFTER MULTIPLE EXPOSURE WITH BTRENGTH PREDICTED FROM SINGLE EXPOSURE CURVES FIGURE 13

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Sequence

● A & A' (5 %ep) ■ B & B' (3 7ter) ● D (2 Step) + B (4 Step) × E (10 Step)

A (3 Step) B (2 Step) B (3 Step) B (3 Step) B (3 Step)

O 

O ▲ (3 Step) ● ▲ (3 Ster

Sequence

COLE:

FIGURE 18 (Cont.) COMPARISON OF STRENGTH AFTER MULTIPLE EXPOSURE WITH STRENGTH PREDICTED FROM SINGLE EXPOSURE CURVES

(c) Stressed Multiple Exposure -- 1.0% Inclustic Strain

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TENSILS YIELD

TENSILE ULTIMATE

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(a) Ultimate Tensile Strength



(b) Yield Strength



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(a) Unstressed Exposures

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## FIGURE 20 STRENGTH DETERIORATION CHARACTERISTICS OF 7075-T6 AT ALL TEST TEMPERATURES

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(b) Stressed Exposures

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FIGURE 21 LARSON-MILLER PARAMETER DIAGRAM

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