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Modeling Fusion of Cellular
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Using Phase Field Theories

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Modeling Fusion of Cellular Aggregates in Biofabrication Using Phase Field Theories

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Abstract

A mathematical model based on a phase field formulation is developed to study fusion of cellular aggregates/clusters. In a novel biofabrication process known as bioprinting [25], live multicellular aggregates/clusters are used to make tissue or organ constructs via the layer-by-layer deposition technique in compatible hydrogels rich in maturogen; the bio-constructs embedded in hydrogels are then placed in bioreactors to undergo the fusion process of self-assembly, maturation, and differentiation to form the desired functional tissue or organ products. We formulate the mathematical model to study the morphological development of the printed bio-constructs during fusion by exploring the chemical-mechanical interaction between cellular aggregates involved. Specifically, we treat the cellular aggregates and the surrounding hydrogels as two immiscible complex fluids and then develop an effective mean-field potential that incorporates the long-range, attractive interaction between cells as well as the short-range, repulsive interaction due to immiscibility between the cell and the hydrogel. We then implement the model using a high order spectral method to simulate the making of a set of tissues/organs in simple geometries like a ring or a sheet of tissues and a Y- or T-shaped vascular junction by the layer-by-layer deposition of spheroidal cellular clusters in the bioprinting technology.

1 Introduction

Tissue fusion is an ubiquitous phenomenon during embryonic development and morphogenesis [30]. The natural tissue fusion in vivo usually occurs in two steps. It starts from the initial tissue opposition and follows by sequential actual tissue fusion after establishing direct contacts between adjacent embryonic tissues. The impaired tissue fusion process during embryonic development resulted in embryonic malformation and defects for example such as cleft palate. In the past, bioengineering processes have been devised to fabricate tissues under controlled conditions using tissue self-assembly, in which one seeds cells into biodegradable polymer scaffolds or gels, which are then cultured in bioreactors for several weeks and finally implanted into the recipient organism, where the maturation of the new organ takes place [17, 24].

In a novel biomimetic biofabrication process, called “bioprinting”, multicellular tissue spheroids or aggregates are used as fundamental building blocks to construct the 3-D tissue or organ [12, 17, 24, 25]. The multicellular aggregates are first prepared in the form of tissue spheroids that consist of thousands of cells blended with bio-compatible hydrogels. They are then directly deposited

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by computer-aided design tools into desired 3-D tissue or organ constructs via the layer-by-layer deposition technique. The bio-constructs immersed in a hydrogel are then placed in bio-incubator for maturation. During the incubating process, the cellular aggregates immersed in compatible hydrogel are expected to fuse into a 3-D tissue or organ following the natural rule of histogenesis and organogenesis [17, 29, 18, 19]. During this process, tissue self-assembly, fusion, differentiation and maturation can occur. Experimental evidence demonstrates that setting lumenized vascular tissue spheroids in a roll, as the result of tissue fusion, a lumenized vascular tube can be formed [25]. Tissue fusion driven is a fundamental biophysical process in emerging organ bioprinting technology.

The bio-constructs ranging from the ones comprised of tissue spheroids to functioning tissues or organs all exhibit fluid behavior during tissue fusion processes and are bona fide soft materials with various degree of viscoelasticity. Numerical simulations of the morphogenesis phenomenon in bioprinting using the Monte Carlo method and experimental evidences [17, 18, 19, 29, 28, 24] pointed out the strong influence of surface tension to the tissue spheroids fusion in the biofabrication process when the biomaterials (tissue spheroids and hydrogels) are regarded as viscous fluids. A separate study using experimental methods and numerical simulation with the cellular Potts model also explored the effect of surface tension and the hydrodynamics to the rounding of cell aggregate [27]. The experimental evidence and the Monte Carlo simulation also demonstrate an interesting compaction phenomenon observed initially in the self-assembly of the bio-construct before it evolves into a smooth tissue or organ [24, 26, 41, 32]. This phenomenon resembles the long range attractive interaction among small particles, but it occurs at a much large length and time scale. The molecular origin of the compaction can be the consequence of cell motility or chemically induced electrostatic interaction between cells mediated by the hydrogel. At large length and time scales, the phenomenon can be modeled using a coarse-grain model featuring the long range aggregate-aggregate interaction, which will be detailed in the next section.

It is known that cell aggregates exhibit fluid-like behavior during the tissue fusion process at large length and time scales [24]. Both the multicellular cell aggregates in the form of tissue spheroids and the hydrogel in which the tissue spheroids are embedded are multiphase complex fluids in nature. Together, they form a binary complex fluid system, in which one fluid is the cellular aggregates and the other is the hydrogel at the coarse-grain level. The dynamics of the binary fluid system can then be modeled in the coarse-grain length scale by models suitable for multiphase complex fluids. Given the large time scale in tissue fusion, elasticity of the tissue spheroids and the bioconstructs made up of the tissue spheroids can be effectively ignored unless an imposed external field is introduced. We therefore, model the bioconstructs at this time scale as viscous fluids.

An excellent review for biofabrication methods, physical mechanisms and the discussion on the fluidity cellular aggregates is available in [24]. An elasto-visco-plastic continuum model is developed to investigate the cell aggregate deformation properties in tension [31], in which a trigger for yield stress is implemented. In large time scale, the model reduces to the viscous limit as well. The biophysical mechanism of tissue fusion process is not completely understood. Modern mathematical modeling and computer simulation opens unique opportunities for explaining physical nature of tissue fusion process and predicting possible undesirable outcomes which is essential for designing optimal bioprinting protocol.

Modeling and simulating immiscible multiphase fluid flows have been challenging both mathematically and technically over the years. Various mathematical theories and computational technologies have been developed to tackle the problem. The front tracking method [10], boundary integral method [16], level-set method [35, 33], volume-of-fluid method [15], immersed-boundary method [34, 23], and phase field method [1, 38, 37, 36] have all been proposed, implemented and refined, each of which has shown effectiveness in designated applications. Some of the methods

are comparable in designing concept while others can be combined to yield more effective computational technology [22]. Among all above, the phase field method for multiphase fluid flows is perhaps the simplest to implement, given that the phase boundary is embedded in a level set of the phase variable governed by a dissipative evolutionary equation, and the most physically relevant. Because of the ease of use, simplicity and relevance to material's physics, refined details in the formulation of the phase field equation can be carefully carved out and the free energy, especially, the interfacial free energy for the multiphase fluid can be devised properly to ensure accuracy in numerical computations and fidelity in physical modeling.

In an immiscible binary fluid, the phase field method employs a phase variable $0 \leq \phi \leq 1$ to track each phase in the binary fluid: $\phi = 1$ describes the phase of fluid I and $\phi = 0$ denotes the one occupied by fluid II while $0 < \phi < 1$ describes the interfacial region. The phase variable is also known as a labeling function for identification purposes. The time evolution of the phase variable ϕ is governed according to the Cahn-Hilliard equation [4, 5]

$$\frac{d\phi}{dt} = \nabla \cdot (\lambda \nabla \mu), \quad (1.1)$$

where λ is proportional to the mobility and μ is the chemical potential of the multiphase fluid system, a functional of the phase variable ϕ , and $\frac{d\phi}{dt}$ is the material derivative. The phase variable ϕ can be identified with the volume fraction of fluid I; so, $1 - \phi$ serves as the volume fraction of fluid II. In this formulation of the transport equation for ϕ , the flux of ϕ is assumed to be proportional to the force due to the prescribed chemical potential. However, a more physically appropriate assumption is to assume the excessive transporting velocity of ϕ , in addition to the bulk/average velocity, is proportional to the force due to the chemical potential, a consequence of the friction dynamics [8, 3]. This leads to the singular or modified Cahn-Hilliard equation

$$\frac{d\phi}{dt} = \nabla \cdot (\lambda_s \phi \nabla \mu), \quad (1.2)$$

where λ_s is the mobility. If we assume mixing only takes place in the interfacial layer, it would be reasonable to assume $\lambda_s = \lambda_s^1(1 - \phi)$, where λ_s^1 is a constant. Hence, the volume fraction will retain a constant within the bulk of each fluid. In the classical Cahn-Hilliard equation, λ is a constant; whereas it is a phase variable dependent function in the modified case. As $\lambda \rightarrow 0$, the transport equation for the phase variable reduces to a pure transport equation for the phase boundary which is commonly used in the level-set method [35, 33].

The phase field model, also known as the diffuse interface method, benefits from the dissipation mechanism in the transport equation for ϕ . There exists an interfacial layer in which two fluids mix due to dissipation. It is expected to maintain a constant value for the phase variable ϕ outside the layer. For a slow varying interface, namely, the interface with a small or intermediate curvature, phase field models yield acceptable interfaces with thickness controlled within a few grid points. When the curvature becomes large, resolving the interfacial layer accurately becomes a challenge. Nonetheless, the material system consisting of cellular aggregates and the host hydrogel matrix does not necessarily maintain a sharp interface at the coarse-grain level given the cell motility. Hence, the phase field method may serve as a good systematic way to model this biomaterial system.

To formulate the cellular spheroid fusion in a hydrogel matrix, we identify the cellular spheroid as fluid I and the ambient fluid, the hydrogel, as fluid II in the framework of phase field theories. We account for three distinct interaction forces between these two “immiscible” fluids. First, we introduce a short-range repulsive bulk potential to maintain the clear separation between the two distinctive fluids, a force to maintain immiscibility between the two fluids; second, we use a configurational entropic term in the interaction potential to describe the philic interaction between

the cellular spheroids in close proximity and the resultant surface tension when coupled with the bulk potential; third, we incorporate a *long-range* attractive interaction potential based on collective Lennard-Jones interaction among the cells physically. The third force is necessary to capture the compaction process observed in the initial self-assembly process of cellular aggregates. Our phase field model is primarily developed based on these interaction forces and their coupling with the mass and momentum transport of the binary fluid system. Should additional forces between the approaching cellular spheroids be identified experimentally at the cellular level, a potential yielding the mean force can be derived accordingly.

We organize the rest of the papers into three sections. In the second section, we derive the detailed phase field model and discuss its nondimensionalization. In the third section, we design an efficient numerical scheme to discretize the governing system of partial differential equations and discuss the numerical implementation that simulates the cellular cluster fusion process to form tissues and organs. Tissues in simple geometries like rings, sheets are simulated along with biofabrication of bifurcating Y or T-shaped vascular veins via layer-by-layer deposition in the fourth section.

2 Mathematical formulation of the phase field model

We treat the cellular cluster as a blob of complex fluids and its surrounding fluid (hydrogel matrix) as a viscous fluid. The cellular cluster along with the surrounding fluid then forms a binary fluid mixture of two immiscible fluids. The binary fluid system is assumed incompressible, which is a reasonable approximation. We use a phase variable ϕ to label each fluid phase in the system:

$$\phi = \begin{cases} 1, & \text{in cellular cluster,} \\ 0, & \text{in host fluid matrix.} \end{cases} \quad (2.1)$$

We denote the average velocity of the fluid mixture by \mathbf{u} . The transport equation for the mass and momentum of the mixture system is governed respectively by

$$\begin{cases} \rho \frac{d\mathbf{u}}{dt} = \nabla \cdot (\phi \tau_1 + (1 - \phi) \tau_2) - (\nabla p + \phi \nabla \mu) \\ \nabla \cdot \mathbf{u} = 0, \end{cases} \quad (2.2)$$

where $\rho = \phi \rho_1 + (1 - \phi) \rho_2$ is the effective density for the binary fluid, ρ_1, τ_1 and ρ_2, τ_2 are the density and the extra stress tensor for the fluid consisting of cellular clusters and the host fluid matrix, respectively, μ is the extended chemical potential of the mixture system, and p is the hydrostatic pressure. The chemical potential of the material system is calculated from the system free energy consisting of the extended “mixing free energy” [9, 13] defined by

$$F_{mix}(\phi) = \frac{1}{2} \int_{\Omega} (\gamma_1 k T |\nabla \phi|^2 + \gamma_2 k T F(\phi)) dx, \quad (2.3)$$

where Ω is the domain that the mixture occupies, $F(\phi) = \phi^2(1 - \phi)^2$ is the Ginzburg-Landau double well potential, k is the Boltzmann constant, T is the temperature, γ_1 is a parameter measuring the strength of the conformation entropy and γ_2 is the strength of the bulk mixing free energy. We introduce an interaction potential due to the long-range cellular interaction:

$$F_{cel}(\phi) = \int_{\Omega} \left(\frac{1}{2} \int_{\Omega} \chi(|x - y|) \phi(x, t) \phi(y, t) dy \right) dx. \quad (2.4)$$

We assume that the long-range cellular cluster interaction is due to the collective attractive interaction between the cells parametrized by the Lennard-Jones potential $\chi(r)$ with

$$\chi(r) = V_{LJ}(r) = 4\epsilon \left(\left(\frac{\sigma}{r} \right)^{12} - \left(\frac{\sigma}{r} \right)^6 \right), \quad (2.5)$$

where ϵ is a suitably chosen energy constant which is influenced by the cellular environment around the surface of the cellular cluster (i.e., it can be a feedback function should more detailed signaling molecular information becomes available when cellular clusters are placed in proximity), σ is the finite distance at which the inter-particle potential is zero and r is the distance between the interacting “particles” (cells) [20]. Like ϵ , σ is a parameter that depends on the detail of cellular interaction when used in this system. Thus, the total free energy of the system is then given by

$$F_{tot} = F_{mix} + F_{cel}. \quad (2.6)$$

The extended chemical potential for the material system is calculated by the variational derivative

$$\mu = \frac{\delta F_{tot}}{\delta \phi} = -\gamma_1 \nabla^2 \phi + \gamma_2 (1 - \phi) \phi (1 - 2\phi) + \int_{\Omega} \chi(|x - y|) \phi(y, t) dy. \quad (2.7)$$

In practice, we use the truncated Lennard-Jones potential

$$\chi(r) = \begin{cases} V_{LJ}(r) - V_{LJ}(r_c) & r \leq r_c, \\ 0 & r > r_c, \end{cases} \quad (2.8)$$

where $r_c \geq 2.5\sigma$. In the calculations we conducted in this paper, we use $r_c = 2.5\sigma$. We remark that the Lennard-Jones potential is attractive at the “long-range” and repulsive in the “short-range”; its use in this model is primarily for the “long-range” attractive effect. The short-range repulsive effect of the Lennard-Jones potential is easily offset by the conformational entropy which promotes a different kind philic interaction at the short to mid-range interaction.

The phase variable ϕ or the volume fraction of the cellular component is transported via the Cahn-Hilliard equation

$$\frac{\partial \phi}{\partial t} + \nabla \cdot (\mathbf{u} \phi) = \nabla \cdot \lambda(\nabla \mu), \quad (2.9)$$

where λ is the mobility that is normally a function of the volume fraction ϕ . Often, it is replaced by a constant for simplicity. At the interface between the two fluid phases, from the Cahn-Hilliard equation, we identify the velocity of the cellular fluid as

$$\mathbf{u}_1 = \mathbf{u} - \frac{\lambda}{\phi} \nabla \mu. \quad (2.10)$$

and that of the host fluid matrix as

$$\mathbf{u}_2 = \mathbf{u} + \frac{\lambda}{1 - \phi} \nabla \mu. \quad (2.11)$$

These two velocities are identifiable only within the interfacial layer between distinctive phases. Within each phase, we assume the complex fluid homogeneous.

The extra stress tensors are given according to the material’s property of each fluid phase involved. Even though the multicellular cluster are Non-Newtonian in nature, the rheological response

in long time is approximately viscous or Newtonian. Considering the large time scale involved in the morphological development process, we assume both fluids viscous with distinct viscosities:

$$\tau_1 = 2\eta_1 \mathbf{D}, \quad \tau_2 = 2\eta_2 \mathbf{D}, \quad (2.12)$$

where η_1, η_2 are the viscosity for the cellular cluster fluid and the host fluid matrix, respectively, \mathbf{D} is the rate of strain tensor for the mixture associated to the average velocity field \mathbf{v} defined by

$$\mathbf{D} = \frac{1}{2}(\nabla \mathbf{u} + (\nabla \mathbf{u})^T).$$

We investigate the interfacial dynamics of the binary fluid associated with the cellular cluster fusion in a 2-D domain $\Omega = [0, H]^2$ and 3-D domain $\Omega = [0, H]^3$. At the boundary of the computed domain $\partial\Omega$, we impose no-flux boundary conditions for the phase variable of the cellular fluid, the Dirichlet boundary condition for the velocity:

$$\begin{cases} \nabla \phi \cdot \mathbf{n}|_{\partial\Omega} = 0, \\ (\mathbf{u}\phi - \lambda \nabla \mu) \cdot \mathbf{n}|_{\partial\Omega} = 0, \\ \mathbf{u}|_{\partial\Omega} = 0. \end{cases} \quad (2.13)$$

We use a characteristic time scale t_0 and a length scale h to nondimensionalize the variables

$$\tilde{t} = \frac{t}{t_0}, \tilde{x} = \frac{x}{h}, \tilde{\mathbf{u}} = \frac{\mathbf{u}t_0}{h}, \tilde{p} = \frac{pt_0^2}{\rho_0 h^2}. \quad (2.14)$$

The length scale h is determined by the computational geometry while the time scale is done by either the growth time scale of the interface or the cellular cluster fusion time scale. The following dimensionless equations then arise

$$\begin{aligned} \Lambda &= \frac{\lambda \rho_0}{t_0}, \quad \Lambda_s = \frac{\lambda_s \rho_0}{t_0}, \quad \Gamma_1 = \frac{\gamma_1 k T t_0^2}{\rho_0 h^4}, \quad \Gamma_2 = \frac{\gamma_2 k T t_0^2}{\rho_0 h^2}, \quad Re_2 = \frac{\rho_0 h^2}{\eta_2 t_0}, \quad Re_1 = \frac{\rho_0 h^2}{\eta_1 t_0}, \\ \tilde{\rho} &= \phi \frac{\rho_1}{\rho_0} + (1 - \phi) \frac{\rho_2}{\rho_0}, \quad \tilde{\epsilon} = \frac{\epsilon t_0^2}{\rho_0 h^2}, \end{aligned} \quad (2.15)$$

where Re_1 and Re_2 are the Reynolds number for the cellular fluid and the ambient fluid, respectively, ρ_0 is an average (or a constant reference) density. For simplicity, we drop the \sim on the dimensionless variables and the parameters. The system of governing equations for the binary fluid in these dimensionless variables are given by

$$\rho \frac{d\mathbf{u}}{dt} = \nabla \cdot (\phi \tau_1 + (1 - \phi) \tau_2) - (\nabla p + \phi \nabla \mu) \quad (2.16)$$

$$\nabla \cdot \mathbf{u} = 0 \quad (2.17)$$

$$\frac{\partial \phi}{\partial t} + \nabla \cdot (\phi \mathbf{u}) = \nabla \cdot (\Lambda \nabla \mu) \quad (2.18)$$

where $\tau_1 = \frac{2}{Re_1} \mathbf{D}$, $\tau_2 = \frac{2}{Re_2} \mathbf{D}$. The dimensionless energy density is now given by

$$f = \frac{\Gamma_1}{2} |\nabla \phi|^2 + \Gamma_2 \phi (1 - \phi) (1 - 2\phi) + \frac{1}{2} \int_{\Omega} \chi(|x - y|) \phi(y, t) \phi(x, t) dy. \quad (2.19)$$

3 Numerical Methods

We apply the phase field model developed in the previous section to study the fusion of the cellular clusters when they are embedded in a host hydrogel matrix, i.e., the time evolution of the overall exterior boundary of the cellular clusters arranged in a designer's pattern for tissue and organ generation. We first describe the numerical scheme to solve the Cahn-Hilliard equation without the velocity (2.18), i.e. zero velocity field. This decoupled system is used to simulate the fusion of spheroids which are in contact initially. For this case, the bulk velocity remains near zero when fusion starts. Thus we only need to focus on the binary fluid evolution without significant fluid motion and employ the Cahn-Hilliard equation to investigate the interfacial dynamics at this moment. Furthermore, we will discuss the numerical scheme to solve the coupled system of Cahn-Hilliard and Navier-Stokes equations (2.16-2.18) to investigate the hydrodynamic effect that drives cellular spheroid fusion when they away from each other initially.

We denote $\Gamma = \Lambda\Gamma_1$ and $\eta = \sqrt{\frac{1}{\Lambda\Gamma_2}}$. The Cahn-Hilliard equation takes the following form:

$$\phi_t + \Gamma\Delta(\Delta\phi - (f'(\phi) + f_l)) = 0, \quad (3.1)$$

where $f(\phi) = \frac{1}{\eta^2}(1 - \phi)(1 - 2\phi)\phi$, $f_l = \frac{1}{\Gamma_1} \int_{\Omega} \chi(|x - y|)\phi(y, t)dy$.

We use the first-order backward Euler method to discretize the time derivative. The numerical scheme reads:

$$\frac{\phi^{n+1} - \phi^n}{\delta t} + \Gamma\Delta\Delta\phi^{n+1} - \frac{\Gamma S}{\eta^2}\Delta(\phi^{n+1} - \phi^n) = \Gamma\Delta(f'(\phi^n) + f_l(\phi^n)), \quad (3.2)$$

where the bilaplace term is treated implicitly and all other nonlinear terms are treated explicitly. An extra term associated with the artificial parameter S is introduced in order to balance the unstable constraint condition on the time step due to the explicit treatment of nonlinear terms [39]. The spatial discretization of the semi-discrete equation is carried out using spectral-Galerkin method with the Legendre-Gauss-Lobatto points in one direction and Fourier method in the other two directions. Hence, in our implementations, the boundary condition in one direction is physical and the other two directions are periodic.

For the coupled system of Cahn-Hilliard equation and the Navier-Stokes equations (2.16-2.18), we use the first-order pressure-correction scheme developed in [6, 7, 40, 11]. For simplicity, we assume the density of the cellular fluid and the hydrogel is the same and then set the density $\rho = 1$ in our nondimensionalization. Assuming that $(\tilde{u}^n, u^n, p^n, \phi^n)$ is known, the numerical scheme reads:

1. *Update of the intermediate velocity field \tilde{u}^{n+1} :*

$$\begin{cases} \frac{\tilde{u}^{n+1} - \mathbf{u}^n}{\delta t} - (\frac{1}{2Re_1} + \frac{1}{2Re_2})\Delta(\tilde{u}^{n+1} - \mathbf{u}^n) + \nabla p^n = N(\phi^n, \mathbf{u}^n), \\ \tilde{u}^{n+1}|_{\partial\Omega} = 0, \end{cases} \quad (3.3)$$

where $N(\phi, \mathbf{u}) = -\mathbf{u} \cdot \nabla \mathbf{u} - \frac{1}{\Lambda}\phi\nabla\mu + (\frac{\phi}{Re_1} + \frac{1-\phi}{Re_2} - \frac{1}{2Re_1} - \frac{1}{2Re_2})\nabla^2\mathbf{u}$.

2. *Enforcing the divergence-free condition through projection to obtain velocity and pressure (u^{n+1}, p^{n+1}) :*

$$\begin{cases} \frac{\mathbf{u}^{n+1} - \tilde{u}^{n+1}}{\delta t} + \nabla(p^{n+1} - p^n) = 0, \\ \frac{\partial(p^{n+1} - p^n)}{\partial n}|_{\partial\Omega} = 0, \end{cases} \quad (3.4)$$

3. Update the phase variable ϕ^{n+1} :

$$\begin{cases} \frac{\phi^{n+1} - \phi^n}{\delta t} + \tilde{\mathbf{u}}^{n+1} \cdot \nabla \phi^n + \Gamma \Delta \Delta \phi^{n+1} - \frac{\Gamma S}{\eta^2} \Delta(\phi^{n+1} - \phi^n) = \Gamma \Delta(f'(\phi^n) + f_l(\phi^n)), \\ \frac{\partial \phi^{n+1}}{\partial n}|_{\partial\Omega} = 0, \frac{\partial \Delta \phi^{n+1}}{\partial n}|_{\partial\Omega} = 0, \end{cases} \quad (3.5)$$

where \mathbf{n} is the outward normal on the boundary. We implemented the coupled system in 2-D and the decoupled system in full 3-D. In the 2-D numerical results presented below, we use 129 Legend basis in one direction and 129 Fourier modes in the other direction. In the 3-D simulations, we use 129 Fourier modes for the extra direction. The computed domain is $[-1, 1] \times [-1, 1]$ for the 2-D case and $[-1, 1] \times [-1, 1] \times [-1, 1]$ for the 3-D case. For the 2-D coupled model (2.16-2.18), we set the parameters in the momentum equations as $Re_1 = Re_2 = 10^{-10}$, $\Lambda = 10^{-7}$. In all simulations, we always fix $\Gamma = 0.0002$, $\eta = 0.01$, $S = 2$, $r_c = 0.5$, $\epsilon = 0.0005$, $\sigma = 0.2$.

4 Results and Discussions

We next investigate cellular spheroid fusion with the numerical code developed in the previous section. We first look into the compaction phenomenon in bioprinting of tissues using the 2-D Cahn-Hilliard/Navier-Stokes solver. In the bioprinting process, when the cellular spheroids are deposited into a designed tissue construct, the cellular clusters undergo an initial compaction before they start fusing. During the compaction process, the center of mass of each cellular spheroid moves towards the geocenter of the construct under an attractive force. This motion occurs on a longer time scale than the fusion process. As an illustration, we will simulate the evolution of a ring shaped bio-construct during the biofabrication process. Then, we simulate the fusion process for the cellular spheroids already in contact using the 3-D Cahn-Hilliard solver for a set of bio-constructs in simple geometries.

4.1 Initial attractive motion of the cellular spheroids in the tissue ring formation

The bio-construct made of the cellular spheroids with the spheroids initially positioned slightly apart undergoes a structural contraction globally. This is shown experimentally in [17] as the compaction phenomenon. We first examine this phenomenon using the 2-D Navier-Stokes/Cahn-Hilliard solver. We investigate how the attractive motion of the cellular spheroids deposited in a ring pattern evolves in time. The lay out of the ring construct consists of 8 cellular spheroids positioned slightly apart. Figure 1 depicts the configurational layout of the 8 cellular spheroids at different time of the simulation. At the initial stage of fusion, the center of mass of each spheroid moves towards the geo-center of the system under the long-range attractive interaction among the cells leading to an evolved ring pattern with the cellular spheroids in contact to each other. As soon as the spheroids get in contact with one another the philic interaction due to the conformational entropy takes over. A smooth tissue ring consisting of the cells forms under the dominating impact of the hydrodynamic surface tension. Given the fact that the volume of the cellular region is conserved in the model, the radius of the smoothed ring keeps shrinking. The radius reduction in time until the spheroids are in contact is recorded in Figure 2, which agrees qualitatively with [26]. The continuing ring shrinking phenomenon during fusion after the cellular spheroids are in contact will be discussed in more details next.

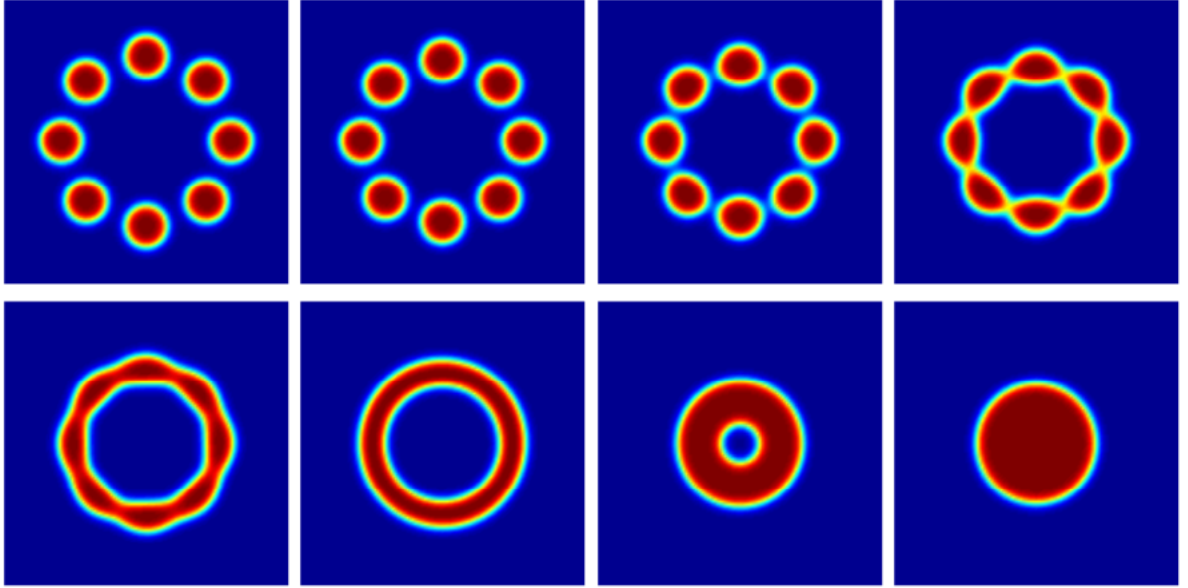


Figure 1: Fusion of a ring construct from ten deposited spheroidal cellular clusters. Initially, the ten spheroids are placed equidistance to the geocenter and completely separated from each other. The long range attractive force moves the center of mass of each spheroid towards the geocenter. Later, surface tension takes over to facilitate the fusion process. The snapshots are for $t = 5, 40, 55, 57, 58, 60, 80, 85$.

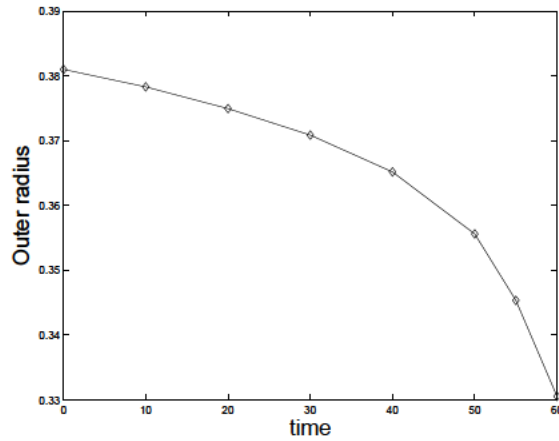


Figure 2: The external radius of the ring construct as a function of time up to $t = 60$. Compaction of the ring construct and later contraction of the smooth ring is evident.

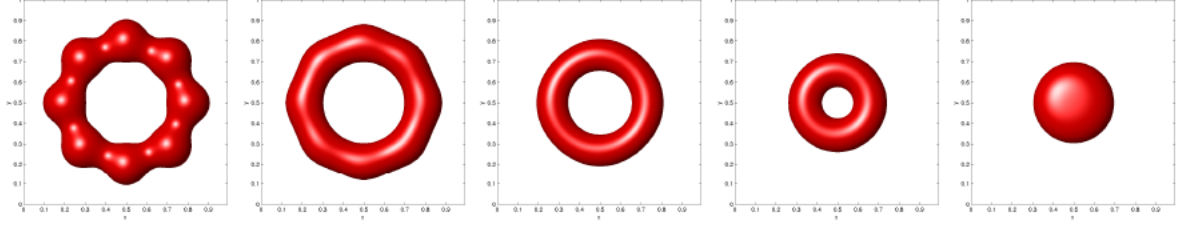


Figure 3: Fusion of a ring construct from eight deposited spheroidal cellular clusters. Surface tension plays a major role in the fusion process. The snapshots are taken at $t = 0.1, 2, 16, 30, 38$.

This simulation demonstrates the effect of the long-range attractive interaction among the cells to the initial configurational evolution of the cellular spheroids within a designed bio-construct. For closely packed spheroids that are already in contact this effect is secondary. The primary force in that situation is the philic conformational entropic one closely related to the surface tension effect. Therefore, in the following simulation, we will instead focus on the interaction dominated by the surface tension in the spheroid fusion process upon contact in 3-D.

4.2 Morphological evolution of a ring construct made of cellular spheroids

Using the 3-D Cahn-Hilliard solver, we investigate how a tissue ring can be fabricated by depositing a ring of cellular spheroids in contact in the bio-printing technology and then watch how this ring can eventually contract into a sphere over a long period of fusion process. The initial deposited cellular spheroids in a construct of the ring shape consists of eight spheroidal cell clusters. Due primarily to the philic effect (dissipation effect) initially, the spheroids join into a corrugated ring; then, the surface tension takes over, the surface is gradually smoothed out. In the meantime the radius of the ring shrinks in order to maintain volume unchanged. After sufficient long time, the ring collapses into a ball. During this time evolution process, the volume of the cellular construct is preserved. Figure 3 depicts five snapshots of the simulation at $t = 0.1, 2, 16, 30, 38$. Compared with the fusion into a smooth ring construct, the evolution to contract into a solid spherical ball takes much longer time shown in Figure 3.

4.3 Tissue sheet fabrication via single layer deposition of cellular spheroids

We then simulate the fabrication of a tissue sheet bio-construct via a single layer cellular spheroids deposition. We lay 9 cellular spheroids initially on a flat surface in a square pattern. The bio-construct gradually joins into a network construct with four visible holes. The entire sheet keeps contracting to close the holes and eventually leads to a solid sheet. The time scale for sealing the holes via fusion is longer than the time used to fusion the spheroids in contact to a smooth networked construct. The dissipation of the cellular material and the surface tension are the major mechanisms driving the fusion process. Figure 4 depicts the simulation all the way to the moment a smooth, solid sheet is formed. The packing of the initial spheroids is not optimal. If we pack the spheroids tightly, the fusion can be reduced by 50% or more as we will shown in the next example.

4.4 Lumenized tube generated via layer-by-layer deposition of cellular spheroids

To simulate the layer-by-layer deposition to fabricate the vascular vein or lumenized tube, we stack rings of spheroids one on top of the other up to three layers in our next simulation. The initially rugged tube-construct evolves into a smooth tube with the walls thickened over time, which

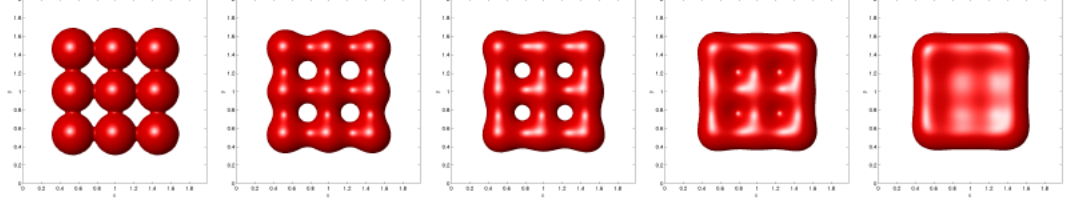


Figure 4: A tissue sheet is fabricated via fusion of a single layer deposited cellular spheroids. The snapshots are taken at $t = 0, 0.5, 1, 2, 3$.

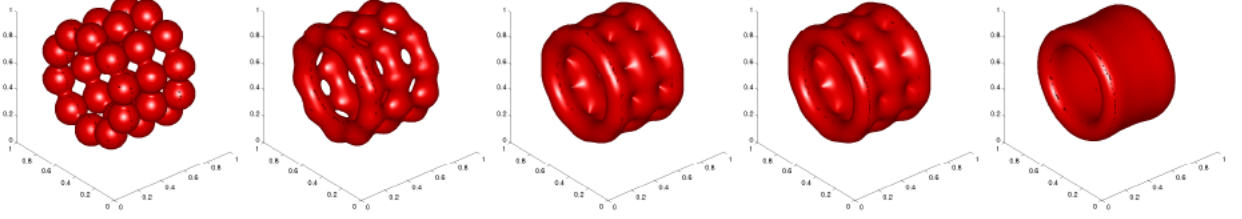


Figure 5: The tube-formation in layer-by-layer deposition of cellular spheroids. Snapshots are taken at $t = 0, 0.1, 0.24, 0.25, 0.5$. It completely fused at $t = 0.25$.

qualitatively captures the morphological evolution of the fusion process observed in experiments (Figure 5).

The positioning or packing of the spheroids relative to each other seems not to make a difference qualitatively. Quantitatively, however, a bioconstruct with an initially more tightly packed spheroids can accelerate the fusion/maturation process significantly. Figure 6 depicts fusion of a tightly packed spheroid tube construct, in which the tube fuses into a hole-less tube in much shorter time than the loosely packed tube construct depicted in Figure 5.

4.5 Bifurcating vascular junction fabrication via layer-by-layer deposition of cellular spheroids

We extend the layer-by-layer deposition strategy to fabricating a bifurcating lumenized tube. If the initial deposition of the construct is done without defect, i.e., too large a hole, both Y-shaped and T-shaped joint can be formed within the same period of time like a single lumenized tube.

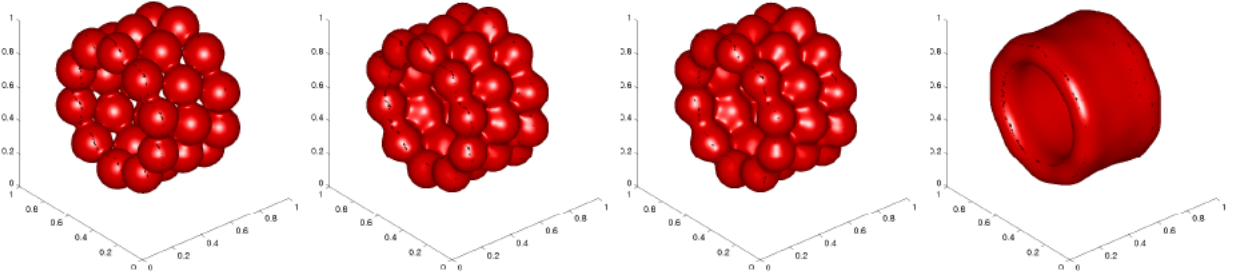


Figure 6: The tube-formation in layer-by-layer deposition of cellular spheroids. Snapshots are taken at $t = 0, 0.008, 0.009, 0.2$. It completely fused at $t = 0.009$.

Figure 7 depicts a Y-shaped bifurcating tube fabricated via fusion of cellular spheroids. Figure 8 portraits a T-shaped bifurcating tube obtained by layer-by-layer deposition of cellular spheroids. If the initial deposition were not done properly, a defective bifurcating tube may form shown in Figure 9, where permanent holes form. Therefore, the initial deposition of the cellular spheroids are instrumental to ensure the quality of the final product. This numerical tool can then be used to study the correlation between the initial configuration of the spheroid arrangement and the possible product defect due to the unexpected holes on the wall. Ultimately, it can be embedded in a computer-aided design software for the bio-printing fabrication.

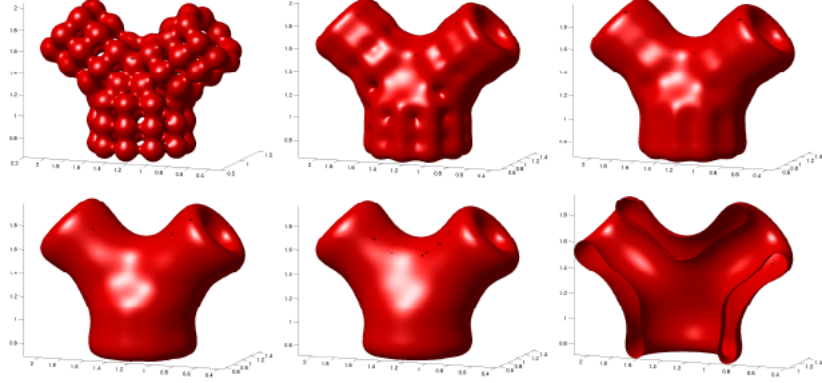


Figure 7: Branching vascular vein fabricated via bioprinting technology. Good branching vascular construct made of layer-by-layer deposition and strategic remedy for the initial profile. Snapshots are taken at $t = 0, 0.2, 0.3, 0.5, 0.8$.

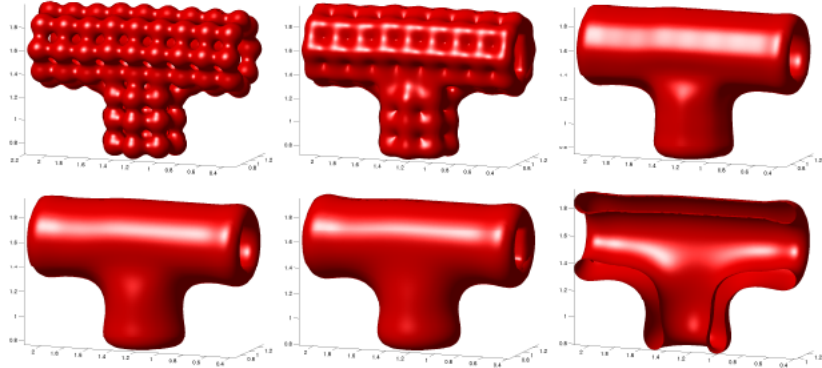


Figure 8: T-shaped branching vascular vein fabricated via bioprinting technology. Perfect branching vascular vein made from layer-by-layer deposition. Snapshots are taken at $t = 0, 0.2, 0.3, 0.5, 0.8$.

5 Conclusion

We have developed a phase field model for studying fusion of cellular aggregates by treating the cellular aggregate and the host hydrogel matrix as two immiscible fluids. The model incorporates the long-range attractive interaction between adjacent cellular aggregates, philic (attractive) interaction in close contact and the repulsive interaction due to immiscibility of distinctive materials at the material's interface. An efficient, high order spectral method is employed to discretize the

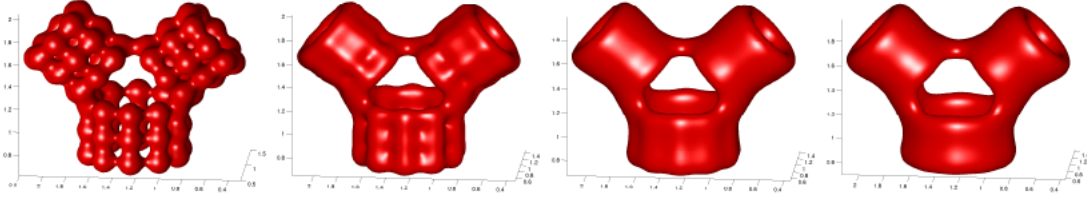


Figure 9: Perforated tube made via layer-by-layer deposition of cellular spheroids with a bad packing method. A defective branching vascular construct forms, in which holes are present in the branching point. Snapshots are taken at $t = 0, 0.2, 0.4, 0.6$.

governing partial differential equation in the model. A numerical solver is developed by implementing the spectral method in 2-D. A 3-D numerical solver is developed based on the Cahn-Hilliard equation only to be applied to investigate the fusion process in which the cellular spheroids are placed in contact. Both solvers are then applied to bio-fabrication of various specific geometric shaped tissues and organs via layer-by-layer deposition of spheroidal cellular clusters. The hydrodynamic effect involving momentum transport is shown to be important in the initial compaction process; it is later dominated by the philic interaction among the cells once the cellular clusters get in contact to each other. The time for fusion of the cellular clusters depends on the initial packing of the cellular spheroids. Tight packing will certainly result in shorter fusion time. The bifurcating bio-constructs also depends on the packing at the turning point. A computer aided design tool is therefore necessary to virtually simulate the entire biofabrication and provides guidance to the bioprinting process in precision.

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