

EFFECT OF STEAM ENVIRONMENT ON CREEP BEHAVIOR OF NEXTEL720/ALUMINA-MULLITE CERAMIC MATRIX COMPOSITE AT ELEVATED TEMPERATURE

THESIS

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Abstract

The tensile creep behavior of an oxide-oxide ceramic matrix composite (CMC) was investigated at 1000 and 1100 °C in laboratory air and steam. The oxide-oxide CMC studied in this research was NextelTM720/alumina-mullite (N720/AM). The composite consists of N720 fibers with 0°/90° fiber orientation and a porous alumina-mullite matrix. There is no interface between fiber and matrix, and the material relies on porosity of matrix for damage tolerance.

Tensile-strain behavior was investigated and tensile properties were measured at 900, 1000 and 1100 °C. The effect of loading rate on tensile properties of N720/AM ceramic matrix composite at 1100 °C in steam was also examined and a strong dependence of tensile behavior on loading rate was observed. Creep-rupture tests were performed at 1100 °C in laboratory air and steam, and at 1000 °C only in steam. Creep run-out time was defined as 100 hours. Tests were performed at the creep stress levels of 109 and 131 MPa in laboratory air and stress levels of 87.5, 109 and 131 MPa in steam at 1100 °C. Creep run-out was achieved at both creep stresses conducted at 1100 °C in laboratory air. At 1100 °C in steam, creep run-out was achieved only at 87.5 MPa. At 1000 °C, tests were performed at the creep stress levels of 131 and 140 MPa only in steam and at both stress levels creep run-out was achieved. Retained strength properties were also measured for test specimens achieving creep run-out of 100 h. Presence of steam caused larger creep strains and the higher stress levels decreased the creep life of the N720/AM ceramic matrix composite. After the mechanical tests, fracture surfaces of the failed specimens were examined using an optical microscope and a scanning electron microscope (SEM).

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Table of	Contents
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Р	'age
Abstract	iv
Acknowledgements	vi
Table of Contents	. vii
List of Figures	ix
List of Tables	lxx
I. Introduction	1
II. Background and Applications	3
 2.1 Ceramic Matrix Composites 2.2 Fibers 2.3 Matrix 2.4 Ceramic Matrix Composite Applications 2.5 Previous Research 2.6 NextelTM 720/AM 2.7 Thesis Objective III. Experimental Arrangements and Test Procedures 	3 8 . 11 . 12 . 16 . 18 . 20 . 20
 3.1 Testing Equipment	. 21 . 21 . 23 . 26 . 29 . 30 . 33 . 33 . 34 . 34
IV. Results and Analysis	. 36

4.1 Section Summary	
4.2 Thermal Expansion	
4.3 Monotonic Tension	
4.4 Creep-Rupture	
4.4.1 Creep-Rupture Tests at 1100°C	
4.4.2 Creep-Rupture Tests at 1000°C	
4.5 Effects of Temperature and Steam on Creep Rupture Behavior	
4.6 Retained Tensile Properties	
4.7 Microstructural Analysis	
4.7.1 Optical Microscopy Analysis	
4.7.2 Scanning Electron Microscopy Analysis	
V. Conclusions	
Appendix. Additional Micrographs	
Bibliography	
Vita	

List of Figures

Figure	Page
Figure 1. Maximum Material Service Temperatures [3].	2
Figure 2. Failure of CMC as a function of interfacial bond strength	6
Figure 3. Typical stress strain curve for CMC with weak interface [8].	7
Figure 4. The classification of fibers [4]	9
Figure 5. As-processed N720/A composite a) overview, optical microscope and b)	porous
nature of the matrix is evident [13].	11
Figure 6. Potential applications of CMCs [14].	14
Figure 7. Ratio of strength to weight as a function of temperature [15]	16
Figure 8. Low magnification views showing dramatic difference between	19
Figure 9. MTS 810 Test Station	21
Figure 10. NESLAB Model HX-75 Chiller.	22
Figure 11. Mechanical Test System	23
Figure 12. Heating Equipment: (a) AMTECO Hot-Rail Furnace	24
Figure 13. MTS Model 409.83B Temperature Controller	25
Figure 14. Susceptor a) front view and b) rear view	26
Figure 15. Zeiss Discovery V12 Optical Microscope	27
Figure 16. FEI FP 2011/11 Quanta 200 3D HV Scanning Electron Microscope	28
Figure 17. Temperature Calibration Specimen.	29
Figure 18. Omega Engineering, Inc, OMNI-CAL-8A-110	30

Figure Page
Figure 19. Uniaxial test specimen (dimensions in mm) [18]
Figure 20. Mitutoyo Corporation Digital Micrometer
Figure 21. Tabbed Test Specimen
Figure 22. Typical tensile test procedure
Figure 23. Typical creep test procedure
Figure 24. CNC Saw
Figure 25. (a) SPI Carbon Coating Machine (b) Carbon-coated failured specimens in the
storage box
Figure 26. Tensile stress-strain curves for N720/AM composite obtained
Figure 27. Tensile stress vs. strain curves for the N720/AM with constant loading rates of
25 and 0.0025 MPa/s at 1100°C in steam. Effect of loading rate on stress-strain
behavior and strength properties is evident
Figure 28. Creep strain vs. time curves for N720/AM composite at 1100°C in air and
steam. Stress levels adjusted for $V_f = 0.44$ are shown in parentheses
Figure 29. Creep strain vs. time curves for N720/A composite at 1100°C in air and steam.
Data from reference [18]
Figure 30. Creep strain vs. time curves for the N720/AM ceramic matrix composite at
1000°C. Stress levels adjusted for $V_f = 0.44$ are shown in parentheses
Figure 31. Creep strain vs. time curves for the N720/A ceramic matrix composite at
1000°C. Data from reference [18]51

Figure 32. Creep strain vs. time curves for N720/AM ceramic matrix composite at 1100	
and 1200°C in laboratory air. Time scale is reduced to clearly. show the creep curve	
at 150MPa. Data at 1200°C from Genelin [16]. Stress levels adjusted for V_f = 0.44	
are shown in parentheses	53
Figure 33. Creep strain vs. time curves for N720/AM ceramic matrix composite at 1000,	,
1100 and 1200°C in steam. Data at 1200°C from Genelin [16]. Stress levels adjusted	1
for $V_f = 0.44$ are shown in parentheses	55
Figure 34. Creep strain vs. time curves for N720/A and N720/AM ceramic matrix	
composites at 150 MPa creep stress level at 1100°C. Stress level adjusted for V_f =	
0.44 is shown in parentheses. Data for N720/A from reference [18]	57
Figure 35. Minimum creep rate as a function of applied stress for N720/AM	58
Figure 36. Minimum creep rate as a function of applied stress for N720/A 5	;9
Figure 37. Creep stress vs. time to rupture for N720/AM ceramic matrix composite 6	50
Figure 38. Effect of prior creep on tensile stress-strain	52
Figure 39. Fracture surfaces of the N720/AM specimens tested in tension to failure in air	r
at: (a) 900 (b) 1000 and (c) 1100°C and (d) 1200°C. Micrograph at 1200°C obtained	ł
from Genelin [22]6	54
Figure 40. Fracture surfaces of the N720/AM specimens obtained in creep tests at	
1100°C in air at: (a-b) 131MPa (equivalent to 150 MPa for V_f = 0.44), t_f = > 100 h	
and (c-d) 109 MPa (equivalent to 125 MPa for $V_f = 0.44$), $t_f = > 100h$	55

Figure 41. Fracture surfaces of the N720/AM specimens obtained in creep tests at 1100°C in steam at: (a-b) 87.5MPa (equivalent to 100 MPa for $V_f = 0.44$), $t_f = >100$ h and (c-d) 109 MPa (equivalent to 125 MPa for $V_f = 0.44$), $t_f = 35.2$ h (e-f) 131 MPa Figure 42. Fracture surfaces of the N720/AM specimens obtained in creep tests in steam at (a-b) 131 MPa (equivalent to 150 MPa for $V_f = 0.44$), $t_f = >100$ h at 1000°C and (b) 131 MPa (equivalent to 150 MPa for $V_f = 0.44$), $t_f = 4.12$ h at 1100°C (c) 136 MPa (equivalent to 150 MPa for $V_f = 0.44$), $t_f = 0.012$ h at 1200°C in steam. Figure 43. Fracture surfaces of the N720/AM specimens subjected to tensile test to failure in steam at 1100°C with the constant loading rates of: (a-b) 25 MPa/s, $t_f = 0.0017h$ Figure 44. Fracture surface of the N720/AM specimens tested in tension to failure 69 Figure 45. Fracture surfaces of specimens tested in tension to failure with displacement control at various magnifications in air at a) 1100°C b) 1000°C and (c-d) 900°C 71 Figure 46. Fracture surfaces of the N720/AM specimens obtained in creep tests at

Figure 48. Fracture surfaces	of the N720/AM	specimens	subjected t	o creep	75
------------------------------	----------------	-----------	-------------	---------	----

Figure 49. Fracture surface of N720/A specimen subjected tensile to failure at 1100°C in
steam with the constant loading rates of: (a-b-c) 25MPa/s and (d-e-f) 0.0025 MPa/s
Figure 50. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 51. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 52. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 53. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 54. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 55. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 56. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 57. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 58 Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air

Figure 59. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	. 85
Figure 60. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	. 86
Figure 61. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	. 86
Figure 62. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	. 87
Figure 63. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	87
Figure 64. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	. 88
Figure 65. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	88
Figure 66. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	. 89
Figure 67. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	89
Figure 68. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	. 90

Figure 69. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	90
Figure 70. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air	91
Figure 71. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	91
Figure 72. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	92
Figure 73. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	92
Figure 74. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	93
Figure 75. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	93
Figure 76. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	94
Figure 77. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	94
Figure 78. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	95

Figure 79. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	. 95
Figure 80. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	. 96
Figure 81. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	. 96
Figure 82. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	. 97
Figure 83. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	. 97
Figure 84. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	. 98
Figure 85. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	. 98
Figure 86. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	. 99
Figure 87. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	. 99
Figure 88. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	100

Figure 89. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	100
Figure 90. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	101
Figure 91. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	101
Figure 92. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	102
Figure 93. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	102
Figure 94. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	103
Figure 95. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	103
Figure 96. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	104
Figure 97. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	104
Figure 98. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	105

Figure 99. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 100. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 101. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 102. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 103. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 104. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 105. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 106. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 107. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 108 Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air

Figure 109. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 110. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air111
Figure 111. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air111
Figure 112. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air112
Figure 113. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 114. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 115. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 116. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air114
Figure 117. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air114
Figure 118. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air

Figure 119. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air 115
Figure 120. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air116
Figure 121. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 122. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 123. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 124. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 125. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 126. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 127. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air 119
Figure 128. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air

Figure 129. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 130. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 131. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 132. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 133. Fracture surface of the N720/AM specimen tested in tension to failure at
1000°C in laboratory air
Figure 134. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 135. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 136. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 137. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 138. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam

Figure 139. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 140. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 141. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 142. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 143. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 144. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 145. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 146. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 147. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 148. Fracture surface of the N720/AM specimen tested in tension to failure with

Figure 149. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 150. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 151. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 152. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 153. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 154. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 155. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 156. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 157. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam

Figure	e 158. Fracture	surface of the	N720/AM s	specimen t	ested in ten	sion to failu	re with
cc	onstant loading	rate of 25MP	a at 1100°C :	in steam			135

Figure 159. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 160. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 161. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 162. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 163. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 164. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 165. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 166. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 167. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 168. Fracture surface of the N720/AM specimen tested in tension to failure with

nstant loading rate of 25MPa at 1100°C in steam

- Figure 170. Fracture surface of the N720/AM specimen tested in tension to failure with
- Figure 171. Fracture surface of the N720/AM specimen tested in tension to failure with
- Figure 172. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam. 142
- Figure 173. Fracture surface of the N720/AM specimen tested in tension to failure with
- Figure 174. Fracture surface of the N720/AM specimen tested in tension to failure with
- Figure 175. Fracture surface of the N720/AM specimen tested in tension to failure with
- Figure 176. Fracture surface of the N720/AM specimen tested in tension to failure with
- Figure 177. Fracture surface of the N720/AM specimen tested in tension to failure with
- Figure 178. Fracture surface of the N720/AM specimen tested in tension to failure with

Figure 179. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 180. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 181. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 182. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 183. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 184. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 185. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 186. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 187. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 0.0025MPa at 1100°C in steam
Figure 188. Fracture surface of the N720/AM specimen tested in tension to failure with

constant loading rate of 0.0025MPa at 1100°C in steam.	15	0
	10	~

Figure 189 Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 0.0025MPa at 1100°C in steam
Figure 190 Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 0.0025MPa at 1100°C in steam
Figure 191 Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 0.0025MPa at 1100°C in steam
Figure 192 Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 0.0025MPa at 1100°C in steam
Figure 193 Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 0.0025MPa at 1100°C in steam
Figure 194 Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 0.0025MPa at 1100°C in steam
Figure 195 Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 0.0025MPa at 1100°C in steam
Figure 196. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 0.0025MPa at 1100°C in steam
Figure 197. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 0.0025MPa at 1100°C in steam
Figure 198. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 0 0025MPa at 1100°C in steam 155

Figure 199. Fracture surface of the N720/AM specimen tested in tension to failure with	h
constant loading rate of 0.0025MPa at 1100°C in steam	156
Figure 200. Fracture surface of the N720/AM specimen tested in tension to failure with	h
constant loading rate of 0.0025MPa at 1100°C in steam.	156
Figure 201. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	157
Figure 202. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	157
Figure 203. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	158
Figure 204. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	158
Figure 205. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	159
Figure 206. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	159
Figure 207. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	160
Figure 208. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	160

Figure 209. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
laboratory air at 1100°C161
Figure 210. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
laboratory air at 1100°C161
Figure 211. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
laboratory air at 1100°C162
Figure 212. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
laboratory air at 1100°C162
Figure 213. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
laboratory air at 1100°C163
Figure 214. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
laboratory air at 1100°C163
Figure 215. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
laboratory air at 1100°C164
Figure 216. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
laboratory air at 1100°C164
Figure 217. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
laboratory air at 1100°C165
Figure 218. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
laboratory air at 1100°C165

Figure 219. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	166
Figure 220. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	166
Figure 221. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	167
Figure 222. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	167
Figure 223. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	168
Figure 224. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	168
Figure 225. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	169
Figure 226. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	169
Figure 227. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	170
Figure 228. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	170

Figure 229. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C17	'1
Figure 230. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C17	'1
Figure 231. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C17	'2
Figure 232. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C17	'2
Figure 233. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C17	'3
Figure 234. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C17	'3
Figure 235. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C17	'4
Figure 236. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C17	'4
Figure 237. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C17	'5
Figure 238. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C17	'5

Figure 239. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	176
Figure 240. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	176
Figure 241. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	177
Figure 242. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	177
Figure 243. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	178
Figure 244. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	178
Figure 245. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	179
Figure 246. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	179
Figure 247. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	180
Figure 248. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	180

Figure 249. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	181
Figure 250. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	181
Figure 251. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	182
Figure 252. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	182
Figure 253. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	183
Figure 254. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	184
Figure 255. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	184
Figure 256. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	185
Figure 257. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	185
Figure 258. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	186

Figure 259. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	186
Figure 260. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	187
Figure 261. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	187
Figure 262. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	188
Figure 263. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	188
Figure 264. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	189
Figure 265. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	189
Figure 266. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	190
Figure 267. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	190
Figure 268. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	191

Figure 269. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	191
Figure 270. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	192
Figure 271. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	192
Figure 272. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	193
Figure 273. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	193
Figure 274. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	194
Figure 275. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	194
Figure 276. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	195
Figure 277. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	195
Figure 278. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	196
Figure 279. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
--	
laboratory air at 1100°C196	
Figure 280. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C197	
Figure 281. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	
Figure 282. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C198	
Figure 283. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C198	
Figure 284. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C199	
Figure 285. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C199	
Figure 286. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	
Figure 287. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	
Figure 288. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1100°C	

Figure 289. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 290. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 291. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 292. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 293. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 294. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 295. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 296. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 297. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 298. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C

Figure 299. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 300. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 301. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 302. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 303. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 304. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 305. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 306. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 307. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 308. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C

Figure 309. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 310. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 311. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 312. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 313. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 314. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 315. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 316. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 317. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 318. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C

Figure 319. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1100°C
Figure 320. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 321. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 322. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 323. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 324. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 325. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 326. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 327. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 328. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C

Figure 329. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 330. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 331. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 332. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 333. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 334. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 335. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 336. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 337. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C
Figure 338. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C

Figure 339. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
steam at 1100°C.	. 227
Figure 340. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	Ļ
steam at 1100°C.	. 227
Figure 341. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	L
steam at 1100°C.	. 228
Figure 342. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	L
steam at 1100°C.	. 228
Figure 343. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
steam at 1100°C.	. 229
Figure 344. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	L
steam at 1100°C.	. 229
Figure 345. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	Ļ
steam at 1100°C.	. 230
Figure 346. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	L
steam at 1100°C.	. 230
Figure 347. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	L
steam at 1100°C.	. 231
Figure 348. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	1
steam at 1100°C	. 231

Figure 349. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
steam at 1100°C	\$2
Figure 350. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
steam at 1100°C	52
Figure 351. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
steam at 1100°C	3
Figure 352. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	3
Figure 353. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	\$4
Figure 354. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	\$4
Figure 355. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	5
Figure 356. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	5
Figure 357. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	6
Figure 358. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	66

Figure 359. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 360. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 361. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 362. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 363. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 364. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 365. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
. Figure 366. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 367. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 368. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C

Figure 369. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 370. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 371. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 372. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 373. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 374. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 375. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 376. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 377. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 378. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C

Figure 379. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 380. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 381. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 382. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 383. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 384. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 385. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 386. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 387. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 388. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C

Figure 389. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	2
Figure 390. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	2
Figure 391. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	3
Figure 392. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	3
Figure 393. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	4
Figure 394. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	4
Figure 395. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	5
Figure 396. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in	
steam at 1100°C	5
Figure 397. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	6
Figure 398. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	6

Figure 399. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 400. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 401. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 402. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 403. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 404. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 405. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 406. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 407. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 408. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C

Figure 409. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	262
Figure 410. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	262
Figure 411. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	263
Figure 412. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	263
Figure 413. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	264
Figure 414. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	264
Figure 415. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	265
Figure 416. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	265
Figure 417. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	266
Figure 418. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	266

Figure 419. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 420. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 421. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 422. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 423. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 424. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 425. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 426. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 427. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 428. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C

Figure 429. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 430. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 431. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 432. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 433. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 434. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 435. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 436. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 437. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 438. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C

Figure 439. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 440. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 441. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 442. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 443. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 444. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 445. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 446. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 447. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C
Figure 448. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in
steam at 1000°C

Figure 449. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	2
Figure 450. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	2
Figure 451. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	3
Figure 452. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	3
Figure 453. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	4
Figure 454. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	4
Figure 455. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	5
Figure 456. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	5
Figure 457. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	6
Figure 458. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	6

Figure 459. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C	7
Figure 460. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	7
Figure 461. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	3
Figure 462. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	3
Figure 463. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C)
Figure 464. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C)
Figure 465. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C)
Figure 466. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C)
Figure 467. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	Ĺ
Figure 468. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	

Figure 469. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C)
Figure 470. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C)
Figure 471. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	;
Figure 472. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	;
Figure 473. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	ŀ
Figure 474. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	ŀ
Figure 475. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	,
Figure 476. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	,
Figure 477. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C)
Figure 478. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C)

Figure 479. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 480. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 481. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 482. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 483. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 484. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 485. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 486. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 487. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 488. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C

Figure 489. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C.	302
Figure 490. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C.	302
Figure 491. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C.	303
Figure 492. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C.	303
Figure 493. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C.	304
Figure 494. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C.	304
Figure 495. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C.	305
Figure 496. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C.	305
Figure 497. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C.	306
Figure 498. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	306

Figure 499. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 500. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 501. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 502. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in
steam at 1000°C
Figure 503. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 504. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 505 Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 506. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air
Figure 507. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air (side view)
Figure 508. Fracture surface of the N720/AM specimen tested in tension to failure at
900°C in laboratory air (side view)

Figure 509. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air (side view)	11
Figure 510. Fracture surface of the N720/AM specimen tested in tension to failure at	
900°C in laboratory air (side view)	11
Figure 511. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	12
Figure 512. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	12
Figure 513. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	13
Figure 514. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air	13
Figure 515. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air (side view)	14
Figure 516. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air (side view)	14
Figure 517. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air (side view)	14
Figure 518. Fracture surface of the N720/AM specimen tested in tension to failure at	
1000°C in laboratory air (side view)	14

Figure 519. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 520. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 521. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 522. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air
Figure 523. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air (side view)
Figure 524. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air (side view)
Figure 525. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air (side view)
Figure 526. Fracture surface of the N720/AM specimen tested in tension to failure at
1100°C in laboratory air (side view)
Figure 527. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam
Figure 528. Fracture surface of the N720/AM specimen tested in tension to failure with
constant loading rate of 25MPa at 1100°C in steam

Figure 529. Fracture surface of the N720/AM specimen tested in tension to f	ailure with
constant loading rate of 25MPa at 1100°C in steam.	
Figure 530. Fracture surface of the N720/AM specimen tested in tension to f	ailure with
constant loading rate of 25MPa at 1100°C in steam	
Figure 531. Fracture surface of the N720/AM specimen tested in tension to f	ailure with
constant loading rate of 25MPa at 1100°C in steam (side view).	
Figure 532. Fracture surface of the N720/AM specimen tested in tension to f	ailure with
constant loading rate of 25MPa at 1100°C in steam (side view).	
Figure 533. Fracture surface of the N720/AM specimen tested in tension to f	ailure with
constant loading rate of 25MPa at 1100°C in steam (side view).	
Figure 534. Fracture surface of the N720/AM specimen tested in tension to f	ailure with
constant loading rate of 25MPa at 1100°C in steam (side view).	
Figure 535. Fracture surface of the N720/AM specimen(2) tested in tension t	o failure
with constant loading rate of 25MPa at 1100°C in steam	
Figure 536. Fracture surface of the N720/AM specimen(2) tested in tension t	o failure
with constant loading rate of 25MPa at 1100°C in steam	
Figure 537. Fracture surface of the N720/AM specimen(2) tested in tension t	o failure
with constant loading rate of 25MPa at 1100°C in steam	
Figure 538. Fracture surface of the N720/AM specimen(2) tested in tension t	o failure
with constant loading rate of 25MPa at 1100°C in steam	

Figure 559. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	330
Figure 560. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	330
Figure 561. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	331
Figure 562. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C	331
Figure 563. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C (side view).	332
Figure 564. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C (side view).	332
Figure 565. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C (side view).	332
Figure 566. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
laboratory air at 1100°C (side view).	332
Figure 567. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	333
Figure 568. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	333

Figure 569. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	334
Figure 570. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C	334
Figure 571. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C (side view).	335
Figure 572. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C (side view).	335
Figure 573. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C (side view).	335
Figure 574. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
laboratory air at 1100°C (side view).	335
Figure 575. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1100°C.	336
Figure 576. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1100°C.	336
Figure 577. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1100°C.	337
Figure 578. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1100°C.	337

Figure 579. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1100°C (side view).	338
Figure 580. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1100°C (side view).	338
Figure 581. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1100°C (side view).	338
Figure 582. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1100°C (side view).	338
Figure 583. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
steam at 1100°C.	339
Figure 584. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
steam at 1100°C.	339
Figure 585. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
steam at 1100°C.	340
Figure 586. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
steam at 1100°C.	340
Figure 587. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
steam at 1100°C (side view).	341
Figure 588. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in	
steam at 1100°C (side view).	341

Figure 589. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C (side view)
Figure 590. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in
steam at 1100°C (side view)
Figure 591. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 592. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 593. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 594. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C
Figure 595. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C (side view)
Figure 596. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C (side view)
Figure 597. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C (side view)
Figure 598. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in
steam at 1100°C (side view)

Figure 599. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	345
Figure 600. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	345
Figure 601. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	346
Figure 602. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C.	346
Figure 603. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C (side view).	347
Figure 604. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C (side view).	347
Figure 605. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C (side view).	347
Figure 606. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in	
steam at 1000°C (side view).	347
Figure 607. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C.	348
Figure 608. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C.	348

Figure 609. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	9
Figure 610. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C	9
Figure 611. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C (side view)	0
Figure 612. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C (side view)	0
Figure 613. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C (side view)	0
Figure 614. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in	
steam at 1000°C (side view)	0

List of Tables

Table Pag	ge
Table 1. Important properties of selected fibers [1,22]	10
Table 2. CMC Applications 1	15
Table 3. Summary of tensile data for N720/AM with 0°/90° fiber	37
Table 4. Summary of creep data for N720/AM with 0°/90° fiber	37
Table 5. Summary of retained properties for N720/AM with	38
Table 6. Summary of thermal properties for N720/AM composite	39
Table 7. Average thermal properties measured for the N720/A and N720/AM composit	es
due to temperature rise from 23°C to	39
900, 1000 and 1100°C. Data for N720/A from Braun[10]	39
Table 8. The average tensile properties for the N720/A and N720/AM composites at	
various temperatures. Data for N720/A from Braun[10,17]	42
Table 9. The average tensile properties for N720/AM composite with constant loading	
rates at 25 and 0.0025MPa/s in steam. Data at 1200°C from Genelin [22]	14
Table 10. Results of Creep-Rupture tests for N720/AM. 4	45
Table 11. Results of Creep-Rupture tests for the N720/A and N720/AM composites with	1
0°/90° fiber orientation at 1100°C. Data for N720/A from Braun[10]	47
Table 12 Results of Creep-Rupture tests for the N720/A and N720/AM composites at	
1000°C. Data for N720/A from Braun [10]	50
Table 13. Retained properties for the N720/AM specimen subjected to prior creep	51

EFFECT OF STEAM ENVIRONMENT ON CREEP BEHAVIOR OF NEXTEL720/ALUMINA-MULLITE CERAMIC MATRIX COMPOSITE AT ELEVATED TEMPERATURE

I. Introduction

Advancing a erospace technology is driving a need for structural materials with ever i ncreasing t hermal capa bilities. For example, all t ypes o f en gines b enefit thermodynamically when materials permit operation at higher combustion temperatures and/or with reduced cooling requirements. Likewise future space/reentry vehicle designs will be nefit g reatly f rom impr oved thermal pr otection systems with the me chanical integrity to serve structural functions [1].

Strength and toughness have a lways be en two major problems in a lot of a reas such a s a erospace i ndustry. Furthermore, w ith t he e ffect of s evere e nvironments including high temperature corrosion, these problems have be come more complicated. The ne cessity for s tructural materials that h ave ex cellent mechanical properties und er extreme conditions have be en raised with developing t echnology. C eramic matrix composites (CMCs) having long-term strength and fracture toughness properties at high temperatures are important materials for such aerospace applications. Additionally, lower densities of C MCs and their high us e temperature, together with a reduced need for cooling a ir, a llow f or i mproved high-temperature performance w hen com pared to conventional nickel based superalloys [2]. Ceramics are the only class of material that
can reliably be used at temperatures above 1100 °C. Figure 1 illustrates the maximum service temperatures of polymers, metals and ceramics.



Figure 1. Maximum Material Service Temperatures [3].

The use of CMCs is not limited to aerospace applications. There are many other areas where CMCs will be of great value, including engine components operating at high temperatures a nd i n c orrosive e nvironments, c utting t ool i nserts, w ear resistant parts, nozzles, exhaust ducts, and energy-related applications [3].

This s tudy i s b ased on the e ffect o f s team e nvironment on c reep be havior of Nextel 720/Alumina-Mullite C eramic M atrix C omposite at el evated temperatures. The

objective is to identify the temperature range where s team environment c auses degradation of c reep r esistance of Nextel720/Alumina-Mullite C eramic M atrix Composite.

II. Background and Applications

2.1 Ceramic Matrix Composites

Ceramics a re d efined a s i norganic, non-metallic ma terials w hich are t ypically crystalline in nature and contain metallic and non-metallic elements such as Al₂O₃, CaO, ZrO₂, S iC, a nd S i₃N₄. There are a s everal broad categories of c eramics c lassifying t he industrial products as follows; clay products, white ware, refractories, cements, abrasives, and advanced ceramics [4].

Advanced ce ramics are m aterials tailored t o posses ex ceptional pr operties (superior m echanical p roperties, corrosion/oxidation r esistance, t hermal, e lectrical, optical or magnetic properties) by controlling their composition and internal structure [4].

Since the beginning of 1990's there has been a great interest in developing a new generation of c eramic composite w hich c an w ithstand high t emperatures i n ox idative atmospheres ov er l ong periods. A further e xpectation i s t hat t he ceramic c omposite products s how inelastic s training, i n ot her terms, display graceful f ailure r ather tha n brittle fracture as ceramics [5]. On the negative side, ceramics tend to be brittle, with low fracture t oughness and damage t olerance. While m etals c an deform pl astically before

fracture, a process that involves extensive energy dissipation, monolithic ceramics do not show signs of plastic deformation and fail in a catastrophic fashion [6].

The major scopes of CMCs can be divided into the areas of [4]:

- biological applications (bioceramics)
- high-temperature applications

Some of these scopes are partly related to each other, and in general they benefit from the low densities of C MCs c ompared to their metallic c ounterparts, leading to lightweight structures [4].

Ceramic ma trix c omposites (CMCs) w ith fiber r einforcement pr ovide t he exemplary way t o r educe t he ne gative ef fects of br ittleness of en gineering ceramics, whilst retaining the further advantageous properties of ceramic structures. The behavior of CMCs is heavily dependent upon t he components used, and a clever combination of reinforcements, interfaces a nd matrix ma terials le ads to sophisticated composites achieving out standing p erformances, especially under s evere environmental c onditions. This e nables t he d esigning e ngineer t o adjust t he c omposites' pr operties directly to various application requirements and load conditions [4].

Besides resistance to oxidation and corrosion, the mechanical properties of CMCs for example fracture toughness and damage tolerance, are of major interest. Thus, those mechanisms which are responsible for such tolerant behavior by CMCs must be adjusted to achieve cr ack de flection and high energy di ssipation in general [4]. In all C MC variations-whether short-or-long-fiber-reinforced-either the interface or the matrix must

meet t his cha llenge. In cons equence, different C MC s trategies ha ve been followed, including weak interface composites (WIC) or weak matrix composites (WMC) [4].

Since the research performed in the early 80's with the first generation of ceramic fibers, CMCs have experienced an exceptional development throughout the last decade, mainly for short terms in the field of aerospace applications (missile, rocket propulsion) where there is a constant need to increase payloads and working temperatures [7].

After the beginning of 1990's, there has been a great interest in developing a new generation of c eramic composite w hich c an w ithstand high t emperatures i n ox idative atmospheres ov er l ong periods. A further e xpectation i s t hat t he ceramic c omposite products s how inelastic s training, i n ot her t erms, di splay graceful f ailure r ather t han brittle fracture as ceramics normally do. A nd, m ore importantly than this, that the high temperature da mage t olerance s hould hol d ove r s everal t housand hour s, s o t hat t he composites e xhibit mi nimized de gradation i n s ervice. Instead, a m oderately hi gh mechanical strength and high toughness in the composites were given greater priority [5].

The fiber/matrix interface affects the behavior of composites. Specially, in CMCs, interfacial bonding affects the fracture behavior of the composite. A strong interfacial bond w ill a llow a noncoming crack go unimpeded through the interface and the composite will fail in a brittle manner. The interaction of a crack in the matrix with a weak interfacial bond, on the other hand, is likely to lead to debonding at the interface through the fibers, followed by crack deflection, crack bridging, fiber fracture and finally fiber pull-out. All these a dditional energy absorbing phenomenal ead to a nenhanced toughness and a non-catastrophic failure mode [3].



Figure 2. Failure of CMC as a function of interfacial bond strength (a) strong interfacial bond; (b) weak interfacial bond [3].

Ceramic Matrix Composites that rely on a weak interface for their toughness can be characterized by their stress strain curve. Initially the stress-strain curve is linear, as the matrix and fibers share the load. As micro-cracks start forming in the matrix, the curve slope starts to decrease. As the cracks in the matrix grow, and then start coalescing, a distinctive knee in the curve is seen. This happens when the cracks propagate through the thickness of the material and all load is transferred to the fibers. At this point the curve is dominated by individual fiber failure and subsequent load transfer to other fibers until the material fails [8]. A typical stress strain curve for a CMC with a weak interface is presented in Figure 3.



Figure 3. Typical stress strain curve for CMC with weak interface [8].

Ceramic matrix composites are designed to minimize the drawbacks of monolithic ceramics. C ompared t o m onolithic c eramics, c ontinuous f iber c eramic c omposites (CFCCs) e xhibit r educed br ittleness a nd de creased s usceptibility t o bot h f laws a nd thermal shock, while maintaining excellent properties at high temperatures [9].

Most important mechanical properties to classify the CMCs are elastic modulus, tensile strength (or flexural strength by bending), fracture toughness, interfacial stresses (thermal and radial stresses) and creep behavior [5]. Creep is deformation of a material over t ime, caused b y a constant or v ery s lightly varying applied load. In the case of ceramics, grain size, porosity, and impurities from processing also contribute to creep [7]. Basically, a s tress/displacement curve contains a 1 inear s tress/strain behavior w hich describes the details of elastic de formation be havior. Then, onc e the stress for matrix cracking has been reached, the C MC shows stress/strain be havior similar to plastic deformation demonstrated in metals [5].

2.2 Fibers

The m ain a dvantage o f C MCs ove r m onolithic c eramics i s t heir superior toughness, tolerance to the presence of cracks and defects, and non-catastrophic mode of failure. It is widely accepted that to avoid brittle fracture behavior in CMCs and improve the damage tolerance, a weak fiber/ matrix interface is needed, which serves to deflect matrix cracks and to allow subsequent fiber pullout [10].

Although the interaction between the fibers, interface and matrix determines the bulk c omposite performance, the fibers are of p articular importance. This is based on several reasons. First, the r einforcing c omponent de termines the m aximum pr operties achievable, and hence fibers are the first choice to develop a CMC. The nature of the fiber influences the production process to be selected and the composite design in terms of i nterface and m atrix t o be i nstalled. S econd, r einforcements of fer various de sign possibilities as, for example, the type of fiber (long, short, filament diameter, aspect ratio, number of filaments per roving, etc.), fiber volume ratio, and the fiber architecture of the composite [4]. In general, fibers can be classified on the basis of their composition and structure that is shown in figure 4.



Figure 4. The classification of fibers [4].

Ceramic fibers are classified as oxide or non-oxide. Non-oxide fibers are made from silicon carbide (SiC). Nicalon, Tyranno, and Sylramic are examples of non-oxide fibers [2]. Table 1 shows important properties of selected oxide ceramic fibers.

Fiber Type	Trademark	Composition	Diameter	Density	Young's
	(Nextel)	(%)	(µm)	(gcm ⁻³)	Modulus
					(GPa)
		62 Al ₂ O ₃			
$Al_2O_3 + SiO_2$	312	24 SiO ₂	8-12	2.7	1.7 / 150
		$14 B_2 O_3$			
$Al_2O_3 + SiO_2$	440	70 Al ₂ O ₃	10-12	3.05	2.1 /190
		28 SiO ₂			
$Al_2O_3 + SiO_2$	550	73 Al ₂ O ₃	10-12	3.0	2.0 / 190
		27 SiO ₂			
		89 Al ₂ O ₃			
$Al_2O_3 + ZrO_2$	650	10 ZrO ₂	11	4.1	2.5 /360
		$1 Y_2O_3$			
$Al_2O_3 + SiO_2$	720	85 Al ₂ O ₃	12	3.4	2.1 / 260
		15 SiO ₂			
		99 Al ₂ O ₃			
a-Al ₂ O ₃	610	0.3 SiO ₂	10-12	3.75	2.6/370
		0.7 Fe ₂ O ₃			

Table 1. Important properties of selected fibers [4].

Reinforcing c omponents c an be pr oduced i n f orm of c ontinuous f ibers, s hort fibers, a nd w hiskers. H igh pe rformance C MCs a re m ainly r einforced b y c ontinuous fibers, since di stinct fi ber a rrangements have lead to explicit c hanges in materials' properties. One of the most important attributes of fibers is their flexibility, this being a function of elastic modulus (E). Hence, reducing the fiber diameter to a sufficiently small value l eads t o very flexible f ibers, even for c eramics w ith very high E and extreme brittleness. T hus, c ontinuous fibers c an b e t ransferred i n s emi-finished f iber pe rforms, which today a major role in the processing of CMCs [4].

The processing of ox ide f ibers be gan dur ing the 1970s, a nd f irst c ommercial products c ontained SiO₂ besides t he m ain c omponent A l_2O_3 [4]. Later, pure α -Al₂O₃ fibers were produced and embedded in alumina matrices in order to provide lightweight

ceramic structures with a high Young's modulus. α -Al₂O₃-fibers with second phases were also developed with the objective to enhance the creep resistance for applications at high temperatures [4, 9, 11].

2.3 Matrix.

The matrix is the continuous phase that provides the shape of the material. The primary functions of the matrix are to transfer load between fibers, separates fibers to prevent adj acent filaments from f ailing, a nd t o pr otect a nd hous e t he f ibers from environmental attack [12].



Figure 5. As-processed N720/A composite a) overview, optical microscope and b) porous nature of the matrix is evident [13].

Although no one ceramic is going to meet all the requirements, one can make a list of some desirable characteristics of a ceramic matrix. That is, what one would like to see in an ideal matrix material. In a real matrix, one can only hope for a large number of these characteristics, An ideal ceramic matrix material should:

• infiltrate a bundle of fibers, whiskers, or particulate perform.

- Form a mechanical or frictional bond with the reinforcement.
- Have no chemical reaction with the fiber reinforcement during fabrication or service.
- No damage to the fiber.
- Have a good resistance to creep, fatigue, and impact.
- Have a high toughness
- Should be chemically stable, i.e., it should be impermeable to moisture, resistant to oxidation, should not hydrate or volatilize, etc [3].

In a ir or ox ygen, ox ide c eramics a re i nherently m ore s table t han no noxide ceramics [3].

2.4 Ceramic Matrix Composite Applications

Today, military and commercial aerospace vehicles desire to go higher, farther, and f aster. This ne ed h as dr amatically i ncreased t he d emand for l ight w eight, hi gh strength structural materials that c an perform in aggressive oper ating environments at extremely high temperatures. Aircraft such as the F-22 and the Boeing 787 D reamliner are pus hing t he t echnological l imits i n or der t o a chieve faster c ruising speeds, l onger operating di stances, and i mproved f light p erformance. "These goals t ranslate i nto material r equirements i nvolving inc reased s trength-to-weight, s tiffness-to-density, a nd improved damage tolerance - all at significantly higher temperatures" [6]

The main advantage of all-oxide composites over non-oxide ones (e.g. SiC/SiC, C/SiC) is their superior resistance to oxidation under typical turbine engine conditions,

since non-oxide fiber-reinforced CMCs show no oxidation resistance at temperatures as far as above 1000°C. A remarkable research effort has been exerted in the development of non-oxide fiber-reinforced CMCs and the resulting composites exhibit high strength, high t oughness i n m any a pplications. H owever, t his i s not t he c ase f or t hose which require oxidation resistance [5].

The application areas of composite are defined as:

- aircraft e ngine c omponents s uch a s t urbine c ombustors, c ompressors and exhaust nozzles.
- 2. ground ba sed ga s t urbine a nd a utomotive components s uch a s combustors, first and second stage turbine vanes and blades
- 3. aerospace engines and missiles and reusable space vehicles and
- 4. industrial appl ications s uch as h eat exchangers a nd r adiant bur ners where pr imarily hi gh t emperature a nd ox idation r esistance of t he material demanded [5].

Besides of fering hi gh te mperature c apability and eliminating c ooling requirements, CMCs offer a significant weight reduction [14]. Figure 6 shows potential applications of CMCs, as non structural and structural parts of aero-engine components.



Figure 6. Potential applications of CMCs [14].

For applications in spacecraft, the major challenge is to provide new materials that will withstand the projected high temperatures and long-term conditions, where metallic or pure carbon materials are insufficiently stable. Moreover, with the availability of flexible manufacturing techniques such as winding, weaving, lay-up laminates and new jointing methods, CMCs become a particularly attractive for these purposes. For the development of new re-entry vehicles such as X38, Hermes, and others, the development of new complex components is vital [4]. However, in space applications, the material life requirements a reless demanding and the environment in space is often non-oxidizing (Table 2.3). Therefore, non-oxide CMCs are hereby more convenient since they will not undergo oxidative embrittlement and display higher mechanical strength [5].

Another field of CMC use in space industry is the development of new propulsion systems. R adiation c ooled noz zle e xtensions a nd c ombustion c hambers a re f avorably fabricated from CMCs, the major advantages being the materials' high strength and light weight, high applicable s ervice t emperatures, and chemical s tability versus liqui d propellant [4]. A ircraft, space and industrial application fields of CMCs are shown in Table 2.

Table 2. CMC Applications

2.1. Aircraft applications [5].

ТҮРЕ	COMPONENT	TYPICAL GOALS
Land based	Combustor	operating at > 1600 °C
gas turbines	Turbine vanes	for 25.000 hrs
Power	Shrouds	900 °C for short
Generation	Combustor	For 25.000 hrs
	Thermophotovoltaic Cells	
Industrial Processing	Chemical Pumps	350 °C
	Gas filters	For 30.000 hrs
	Furnace hardware	in chemical hazardous
		environment

2.2. Industrial and power generation applications [5].

AIRCRAFT TYPE	COMPONENT	TYPICAL GOALS
Civil Aircraft	Compressor	For both components
Gas Turbines	Combustor turbine	> 1300 °C for > 10.000
		hrs
Commercial	Combustor	> 1600 °C
Supersonic Transport	Exhaust Nozzle	>800 °C for 10.000 hrs
Military	Combustor turbine	For both components
	Exhaust nozzle	> 1300 °C for over 1000
		hrs

2.3. Space applications [5].

ТҮРЕ	COMPONENT	TYPICAL GOALS
Missile	Combustor	Operating
	Turbine rotors	at > 1400 °C
Space Vehicles	Turbomachinery	Very high temperatures
	Nozzles	> 1600 °C for short
	Thrust chambers	periods of time
Satellites	Maneuvering Thrusters	> 1700 °C for > 10 hrs

As s een in figure 7, ceramic m atrix com posites ar e l ess de nse t han high temperature s uperalloys y et s till ha ve c omparable s trength to weight r atios and much greater temperature operating ranges [15].



Figure 7. Ratio of strength to weight as a function of temperature [15].

2.5. Previous Research

There a re s ome pr evious r esearch efforts ex ist on the N extelTM 720/Alumina-Mullite and NextelTM720/Alumina c omposites with $0^{\circ}/90^{\circ}$ fiber or ientation in both a ir and steam environments at AFIT.

Genelin [16] s tudied t he t ensile creep b ehavior of t he N 720/Alumina-Mullite ceramic matrix composite with 0°/90° fiber orientation at 1200 °C. Creep tests performed in air, steam and argon which is a non-oxidizing environment. He first investigated the stress-strain behavior of the composite. Additionally, the influence of the loading rate on tensile behavior of the material was explored. Creep tests were conducted at 1200 °C in

air, steam and argon. According to his results, the N720/AM composite exhibits primary and secondary creep regimes in air and steam. The largest creep strains were accumulated in steam. He observed that the presence of steam dramatically reduced creep lifetimes of the N720/AM and he also found that the presence of argon had a detrimental effect on creep performance of the N720/AM.

Harlan [17] examined creep-rupture be havior of t he N 720/Alumina c eramic matrix c omposite w ith 0/ 90° f iber or ientation i n bot h l aboratory air a nd s team environments at 1200 a nd 1330 °C. Creep stress levels ranging from 80 t o 154 M Pa at 1200 °C and creep s tress l evels of 50 a nd 10 0 M Pa a t 1330 °C w ere i nvestigated. According to his results, the material performed well during creep tests in air, but the material's creep performance in steam degraded at both temperatures.

Braun [18] studied the creep behavior of the N720/A composite with 0/90° fiber orientation. The creep behavior of the N720/A composite was characterized in air and in steam for different creep stress levels at 1000 and 1100 °C. According to his results the tensile properties were strongly influenced by the loading rate at 1100 °C in steam. Braun found that the N720/A ceramic matrix c omposite exhibits de creased da mage t olerance and reduced creep lifetimes at t emperatures ≥ 1100 °C. H e explored the i nfluence of loading rate at 1100 °C. The results of his study demonstrated that at temperatures ≥ 1100 °C t he pr esence of s team caus es de gradation of cr eep resistance of t he N 720/A composite, as manifested by shortened creep lifetimes [18, 19].

Siegert [20] investigated the creep behavior of the N720/A composite with $\pm 45^{\circ}$ fiber orientation. In his effort, creep tests were performed in air, steam and argon. The

results showed that specimens tested in air produced lower creep rates than specimens tested at the same stress levels in other environments. He reported that environment did not appear to have a significant influence on the creep life of $\pm 45^{\circ}$ specimens.

2.6. NextelTM 720/AM

The N720/AM is an oxide-oxide ceramic matrix composite consists of N720 fibers with 0°/90° fiber orientation and a porous alumina-mullite matrix. There is no interface between fiber and matrix, and the material relies on porosity of matrix for damage tolerance.

Fiber that exists in NextelTM720/Alumina-Mullite ceramic matrix composite is manufactured by Minnesota Mining and Manufacturing Company ($3M^{TM}$). This fiber is composed of 85%Al₂O₃ and 15%SiO₂ in the form of α -alumina [21].

Early investigations on alumina and mullite based fibers (e.g. NextelTM312, 450 and 550) s howed t hat t hese f ibers de grade i n strength above 1250 °C on l ong-term exposure (> 100 h). The loss of strength may arise from grain growth within the fiber and/or from chemical reaction of fiber with the matrix or interface material under high-temperatures. With NextelTM720, the first p rogress is r eported. The superiority of t his fiber relies on the grain growth inhibition of alumina, achieved by addition of mullite [5].

Mullite is a live compound of a lumina and silica in the compositional range of 71-75 % a lumina. C ommonly mullite is r epresented by formula, 3Al₂O₃.2SiO₂. It has excellent strength and creep resistance as well as low thermal expansion and conductivity [3].

The fiber NextelTM720 shows the highest thermal stability in the Nextel family. The fiber consists of a lumina a nd m ullite g rains, a nd i s m anufactured via a s ol-gel process. Small plate-like Al_2O_3 grains (70-100 n m) are distributed be tween and inside mullite grains (300-500 nm), which consist of smaller subgrains. At 1400 °C, 85% of the room-temperature strength is retained and the creep resistance is superior to that of all other Nextel fibers [4].



Figure 8. Low magnification views showing dramatic difference between a) fibrous failure normally seen with porous matrix composites and b) a brittle failure [1].

The composite manufactured by COI Ceramics (San Diego, CA), was supplied in a form of a 3.2 mm thick plate, comprised of 12 0°/90° woven layers [22] and the N720/Alumina-Mullite specimens used in this research had a fiber volume fraction of 38.5%.

2.7. Thesis Objective

The previous study revealed degrading effect of steam environments on m aterial performance unde r bot h s tatic a nd c yclic l oadings. A dditional m atrix sintering w as observed at 1200 °C, which l ed t o t he l oss o f m atrix por osity a nd d eterioration of composite toughness. This degradation process was accelerated in the presence of steam. In a ddition, t he p rior study r evealed t hat at 1200 °C t he creep performance of NextelTM720/Alumina-Mullite C MC, a ma terial s ystem w hich relies on a por ous alumina/mullite matrix for damage tolerance, also deteriorates drastically in the presence of steam [16, 22]. The objective of this study is to identify the temperature range where steam environment causes degradation of creep resistance of the NextelTM720/Alumina-Mullite CMC.

III. Experimental Arrangements and Test Procedures

This c hapter e xplains bot h t he t esting a nd s upporting e quipments us ed i n t his research. A dditionally, detailed de scriptions of all t est procedures a nd microstructural analysis are presented.

3.1 Testing Equipment

3.1.1 Mechanical Equipments

A servo-hydraulic Material T est S ystems (MTS) 810 machine was us ed for all tensile, creep and stress rate tests. The maximum loading capacity was 5.5 kip (25 kN). This vertically actuated machine is shown in Figure 9.



Figure 9. MTS 810 Test Station.

Water-cooled hydraulic w edge grips w ith surfalloy s urfaces grip the te st specimens. An MTS T est S tar II di gital c ontroller w as us ed f or t est c ontrol a nd da ta acquisition.

A grip pressure of 8 M pa, prevents specimen slippage while keeping the gripped portion of the specimen from damage, was used in all tests. A NESLAB model HX-75 chiller provided 15 °C deionized water circulation through the wedges which ensured the grips to be cooled. It is shown in Figure 10.



Figure 10. NESLAB Model HX-75 Chiller.

Force measurement was acquired by a 25 kN (5.5 kip) maximum capacity MTS Force Transducer (Model 661.19E-04). A uniaxial, high-temperature, low contact force MTS Extensometer (Model 632.53E-14) with the 12.5 mm gage length performed the strain measurement. Displacement measurement was obtained by an LVDT internal to the MTS servo-hydraulic machine. Figure 11 shows the mechanical test system includes close-up view of the transducer, top wedge grips, and extensometer set-up.



Figure 11. Mechanical Test System.

3.1.2 Environmental Equipments

To maintain the elevated temperature and environment required for testing, the mechanical t esting s tation w as e quipped w ith a dual z one A mteco H ot R ail F urnace System. The dual z one Amteco H ot R ail F urnace is shown in Figure 12a. Two heating elements, internally insulated with alumina, were established each sides of the furnace. The i nsulation w as m odified t o e nable t he ov en t o c lose around t he t est s pecimens without any undesirable interference. Figure 12b shows the modified furnace insulation where he ating elements and control thermocouple are visible. An R-type thermocouple,

fitted on each side of the furnace, provided temperature information of the chamber to the controller unit.







Figure 12. Heating Equipment: (a) AMTECO Hot-Rail Furnace (b) Modified furnace insulation with heating elements and control thermocouple. (c) Heating Element

The dual zone Amteco Hot Rail Furnace System (see Fig. 12) was controlled by an MTS Model 409.83B Temperature Controller that is shown in Figure 13.



Figure 13. MTS Model 409.83B Temperature Controller.

Continuous s team e nvironment w as obt ained by an Amteco HRFS-STMGEN Steam Generation System during the tests conducted in steam. Steam Generation System supplied de -ionized w ater us ing a o ne-gallon w ater r eservoir. A n alumina s usceptor, cylinder tube with e nd c aps, which fits inside the furnace provided a positive pressure chamber around the specimen forcing out the dry air. The gauge section of the specimen was placed inside the susceptor while the ends passing through the slots. Note that the susceptor was not used during the tests in air. Figure 14 shows the front and rear view of the s usceptor with the entrance slot for specimens, and holes for the extensometer and steam tube.



Figure 14. Susceptor a) front view and b) rear view

3.1.3. Microstructural Characterization Equipments

Both an optical microscope and a scanning electron microscope (SEM) were used for post test analysis of the failed specimens' fracture surfaces. A Zeiss Discovery V12 optical microscope equipped with a Z eiss A xioCam H Rc di gital cam era w as us ed to examine the da mage z ones of the specimens. Figure 15 shows a Zeiss Discovery V12 optical microscope. Micrographs were taken at various magnifications up to 100X by the optical microscope.



Figure 15. Zeiss Discovery V12 Optical Microscope.

An F EI FP 2011/11 Q uanta 200 HV S canning Electron M icroscope s hown in Figure 16 was us ed to examine t he s pecimen microstructure characterization at magnifications of up to 20,000X.



Figure 16. FEI FP 2011/11 Quanta 200 3D HV Scanning Electron Microscope.

The scanning electron microscope (SEM) is a type of electron microscope that images the sample surface by scanning it with a high-energy beam of electrons in a raster scan pattern. The electrons interact with the atoms that make up the sample producing signals that contain information about the sample's surface topography, composition and other properties such as electrical conductivity. The SEM produces i mages by probing the specimen with a focused electron beam that is scanned across a rectangular area of the specimen. At e ach point on t he specimen the incident electron be am loses s ome energy, and that lost energy is converted into other forms, such as heat, emission of lowenergy secondary electrons, light emission (cathodoluminescence) or x-ray emission. The signals result from interactions of the electron beam with atoms at or near the surface of the sample. The display of the SEM maps the varying intensity of any of these signals into the image in a position corresponding to the position of the beam on the specimen when the signal was generated. The SEM micrographs have a v ery large depth of field yielding a cha racteristic t hree-dimensional ap pearance us eful f or understanding t he surface structure of a sample [23].

3.2 Test Procedures

3.2.1 Temperature Calibration

To obtain the desired test temperature of the specimen, the furnace temperature controller was calibrated prior to mechanical testing. Calibration was performed by using a test specimen fitted w ith 2 R-type t hermocouples shown in F igure 17. T he thermocouples were wrapped with a high temperature wire to ensure contact with the specimen. This specimen was placed into the M TS machine following the same procedures during tests.



Figure 17. Temperature Calibration Specimen.

After placing the calibration specimen, the furnace temperature was raised until the specimen temperature reached the desired temperature level at a rate of 1 °C/s. To obtain the pr ecise el evated temperature cal ibration, the t hermocouple t emperature readings were closely monitored while the oven temperatures were manually increased to establish the desired temperature of the test specimen. The temperature of the specimen was read by a portable temperature reader shown in Figure 18.



Figure 18. Omega Engineering, Inc, OMNI-CAL-8A-110 Portable Temperature Reader

3.2.2 Preparation of Mechanical Testing

Prior to mechanical testing, specimen fabrication process was completed. Specimens were cut from the composite panel by the AFIT machine shop technicians using a high pressure water-jet machine. The panel was sandwiched between two thin aluminum sheets to prevent possible edge wear. Water-jet machine has a computercontrolled nozzle that sprays water mixed with garnet particles at high pressures to precision-cut different kinds of materials. After cutting process, three steps of cleaning process were followed to remove the debris from the water-jet process. The cut specimens were placed into an ultrasonic bath and exposed to de-ionized water for 20 min, then soaked in 200-proof ethyl alcohol for 20 min. The cleaning process was completed after drying the specimens in the Omegalux LMF-3550 Benchtop Muffler Furnace at 250 °C for at least 2 hours. Figure 19 shows the uniaxial test specimen geometry.



Figure 19. Uniaxial test specimen (dimensions in mm) [18].

Servo-hydraulic testing machine was warmed up in order to ensure the hydraulic fluid was at its operating temperature. The Function Generator mode of the MTS 810 Machine was us ed to c ycle the actuator in displacement control for a minimum of 30 minutes. In this warm up process, square wave function with the frequency of 3 Hz and amplitude of 0.01 inches was used.

The sizes (width and thickness) of the specimen gauge section were measured at least three times by a Mitutoyo Corporation Digital Micrometer (Model N TD12-6"C) shown in figure 20 and an average was taken.



Figure 20. Mitutoyo Corporation Digital Micrometer

The cross-sectional area of the specimen was calculated using the measurements. The cal culated cross-sectional a rea w as us ed t o f ind t he l oad ne eded t o a chieve t he desired test stress level, using the equation:

$$P = \sigma x A$$

where *P* is load in Newtons (N), σ is stress in Pascals (Pa), and A is cross-sectional area in meters squared (m²).

To protect the specimen from damaging by the pressure of the grips, the edges of the specimen were tabbed with rectangular fiberglass papers. (see Figure 21)



Figure 21. Tabbed Test Specimen.

3.2.3 Monotonic Tensile Tests

All monotonic tensile tests to failure were conducted at 900, 1000 and 1100 °C in laboratory air . The specimens were heated under zero load from room temperature to the elevated temperature at 1 °C/s and held at that temperature for 20 minutes. After the 20 minutes of dwell time, the specimen was loaded using displacement control at a constant rate of 0.05mm/s until failure. Load, strain, displacement, time data were recorded every 0.05 s from the beginning to the failure for all tensile tests.

Туре	Name	Start	Interrupt
Ð.	Record Warm Up	<procedure>.Start</procedure>	Ramp Ovens to Temp.Done
	Ramp Ovens to Temp	<procedure>.Start</procedure>	
Ð.	Record Tensile Data	Ramp Ovens to Temp.Done	Shut Off Ovens.Done
	Load to Failure (Dsp Ctl)	Ramp Ovens to Temp.Done	
-	Shut Off Ovens	Load to Failure (Dsp Ctl).Done	

Figure 22. Typical tensile test procedure.

3.2.4 Creep-Rupture Tests

Creep-rupture tests were conducted in load control at 1000 °C only in steam and at 1100 °C in both laboratory air and steam . C reep run-out was defined as 100 h a t a given creep stress for all creep rupture tests. Specimens that achi eved run-out in creep tests were unloaded to zero load and then they were subjected to tensile tests to failure in order to evaluate the retained tensile properties

Similar pr ocedures w ere f ollowed during t he creep tests i n steam exc ept t he addition of an alumina susceptor. Steam was pumped through a feeding tube into the rear section of the susceptor at a rate of 30 mL/minute expelling the dry air from the susceptor

and c reating a n ear 100 % s team environment i nside t he s usceptor. Figure 23 shows a typical creep test procedure.

Туре	Name	Start	Interrupt
$\mathbf{\Sigma}$	Upper/Lower Disp Limits	<procedure>.Start</procedure>	
₽ ₽	Record Warm Up	<procedure>.Start</procedure>	Warm Up/Hold Ovens.Done
	Warm Up/Hold Ovens	<procedure>.Start</procedure>	Upper/Lower Disp Limits.Done
₽ ₽	Record Load Up	Warm Up/Hold Ovens.Done	Ramp Up (Load Ctrl).Done
\square	Ramp Up (Load Ctrl)	Warm Up/Hold Ovens.Done	Upper/Lower Disp Limits.Done
t.	Record Creep (0-5 min)	Ramp Up (Load Ctrl).Done	
t.	Record Creep (5-10 min)	Record Creep (0-5 min).Done	
Q.	Record Creep (10 min -1 hr)	Record Creep (5-10 min).Done	
Ø.	Record Creep (1-3 hr)	Record Creep (10 min -1 hr).Done	
Ø.	Record Creep (3-5 hr)	Record Creep (1-3 hr).Done	
₽ ₽	Record Creep (5-25 hr)	Record Creep (3-5 hr).Done	
₽ ₽	Record Creep (25-100 hr)	Record Creep (5-25 hr).Done	Hold Load 100 Hrs (Load Ctrl). Done
\square	Hold Load 100 Hrs (Load Ctrl)	Ramp Up (Load Ctrl).Done	Upper/Lower Disp Limits.Done
₽ ₽	Record Ramp Down	Hold Load 100 Hrs (Load Ctrl).Done	Ramp Down (Load Ctrl) Done
\square	Ramp Down (Load Ctrl)	Hold Load 100 Hrs (Load Ctrl).Done	Upper/Lower Disp Limits.Done
₽ ₽	Record Tensile Test	Ramp Down (Load Ctrl).Done	Tensile Test (Dsp Ctrl).Done
\square	Tensile Test (Dsp Ctrl)	Ramp Down (Load Ctrl),Done	Upper/Lower Disp Limits.Done
	Shut Down Ovens	Tensile Test (Dsp Ctrl).Done	

Figure 23. Typical creep test procedure.

3.2.5 Tensile Tests at Stress Rates 0.0025 and 25 MPa/s

To observe the effects of both slow and fast stress rate during tensile, two repeated tests were conducted in force control at 1100 °C in steam environment. The tests were conducted at the stress rates of 0.0025 and 25 MPa/s.

3.2.6 Microstructural Characterization

Optical microscope and SEM were used for post-test microstructure examination. Various magnifications were used to examine the entire fracture surface of each half of the failed specimens.

Note that, one optical micrograph was taken per side of failed half specimens. After that, only one half of the each failed specimens was prepared to get a clear view in the S EM. D uring t his pr eparation process, failed specimens w ere mounted on t o aluminum platforms by gluing with silver paste a fter the y h ad been cut by C NC s aw shown in F igure 24. The other half parts of the failed specimens were collected for documentation.



Figure 24. CNC Saw.

SEM, ut ilizing the pr ocess de scribed i n c hapter 3.1.3, r equires c onductive materials t hat pr event cha rging i n order t o enable cr eating cl ear i mages. A non-conductive s pecimen cause a cha rge w hich obscures r esolution capability a nd may damage the equipment. Since the CMCs are non-conductive materials, they were first coated w ith carbon us ing an S PI-Module C ontrol a nd C arbon C oater t o obs erve t he fracture surfaces of the specimens. The SPI-Module Control and Carbon Coater and the storage box includes carbon coated samples are shown in Figure 25.



Figure 25. (a) SPI Carbon Coating Machine (b) Carbon-coated failured specimens in the storage box.

IV. Results and Analysis

4.1 Section Summary

This chapter explains the detailed experimental results of the all tests were performed in this research.

First, the chapter starts with the presentation and discussion about thermal properties of the N720/AM ceramic matrix composite. Then, the results of tensile tests with both displacement and constant loading rate control are presented. Next, the results of creep-rupture tests are explained and discussed. Effect of temperature and steam environment on creep-rupture behavior is widely discussed. Comparisons of the results from this research and previous efforts are also presented. Finally, fracture surfaces of failed N720/AM specimens are characterized. Micrographs obtained by using optical and scanning electron microscope are discussed. Following tables represent the summary of

the experimental tests performed in this research.

obtained in tensile tests.							
Specimen	Control Method	Environment	Temperature	Elastic	UTS		
(#)			(°C)	Modulus	(MPa)		
				(GPa)			
1	0.05mm/s	Air	900	67.5	146.4		
2	0.05mm/s	Air	1000	65.5	161		
4	0.05mm/s	Air	1100	63.5	160.2		
13	25 MPa/s	Steam	1100	65.5	150		
14	0.0025 MPa/s	Steam	1100	44.0	132		
15	0.0025 MPa/s	Steam	1100	44.2	130		
16	25 MPa/s	Steam	1100	62.3	155		

Table 3. Summary of tensile data for N720/AM with 0°/90° fiber orientationobtained in tensile tests.

Table 4. Summary of creep data for N720/AM with 0°/90° fiber orientation obtained in creep-rupture tests.

Specimen	Environment	Temperature	Elastic	Creep	Creep	Creep
(#)		(°C)	Modulus	Stress	Strain	Life
			(GPa)	(MPa)	(%)	(h)
6*	Air	1100	65.2	109	0.2	>100
7*	Air	1100	64.7	131	0.23	>100
8	Steam	1100	62.9	131	0.42	4.12
9	Steam	1100	66.3	109	0.87	35.2
10*	Steam	1100	64.4	87.5	0.69	>100
11*	Steam	1000	65.4	131	0.20	>100
12*	Steam	1000	65.3	140	0.21	>100

* Run-out (100h)
| Specimen
(#) | Temperature
(°C) | Environment | Creep
Stress
(MPa) | Retained
Modulus
(GPa) | Retained
Strength |
|-----------------|---------------------|-------------|--------------------------|------------------------------|----------------------|
| 11 | 1000 | Steam | 131 | 58.7 | 166 |
| 12 | | Steam | 140 | 60.6 | 173 |
| 6 | | Air | 109 | 59.9 | 168 |
| 7 | 1100 | Air | 131 | 55.8 | 174 |
| 10 | | Steam | 87.5 | 55.4 | 162 |

Table 5. Summary of retained properties for N720/AM with 0°/90° fiber orientation subjected to prior creep.

4.2 Thermal Expansion

The coefficient of linear thermal expansion was calculated by using thermal strain measured in the tests. Thermal strain was measured using the linear part of the strain-time curve while heating up the test environment to the elevated test temperature. The following equation gives us the relation between thermal strain and the coefficient of linear thermal expansion

$$\varepsilon_t = \alpha_t \cdot \Delta T$$

where ε_t is the thermal strain (m/m), α_t is the coefficient of linear thermal expansion and ΔT is the change in temperature (K) that was the difference between the elevated temperature and the room temperature. The room temperature was defined as 23 °C for all tests performed in this effort. Thermal properties are summarized in Table 6.

Specimen	Environment	Temperature	Thermal	Coefficient of Linear				
(#)		(°C)	Strain	Thermal Expansion				
			(%)	$(10^{-6} K^{-1})$				
1	Air	900	0.56	6.39				
2	Air	1000	0.69	7.07				
4	Air	1100	0.71	6.56				
6	Air	1100	0.69	6.38				
7	Air	1100	0.70	6.49				
8	Steam	1100	0.74	6.86				
9	Steam	1100	0.72	6.67				
10	Steam	1100	0.71	6.58				
11	Steam	1000	0.66	6.78				
12	Steam	1000	0.63	6.45				
13	Steam	1100	0.73	6.76				
14	Steam	1100	0.70	6.49				
15	Steam	1100	0.76	7.06				
16	Steam	1100	0.74	6.88				

 Table 6. Summary of thermal properties for the N720/AM composite obtained from all tests.

The average values of ε_t at 900, 1000 and 1100 °C were 56%, 0.66% and 0.72%

and the average of calculated α_t values were 6.39, 6.76 and 6.67 10^{-6} K⁻¹, respectively.

Average thermal measurements for the N720/A and N720/AM ceramic matrix

composites are summarized in Table 7.

Table 7. Average thermal properties measured for the N720/A and N720/AM
composites due to temperature rise from 23°C to 900, 1000 and 1100 °C. Data for
N720/A from Braun[18]

Material	Temperature	Average Thermal Strain (%)	Coefficient of Linear Thermal Expansion (10 ⁻⁶ K ⁻¹)
N720/A	1000	0.58	7.57
	1100	0.68	8.2
	900	0.56	6.39
N720/AM	1000	0.66	6.76
	1100	0.72	6.67

The average α_t for the N720/AM composite due to temperature rise from 23 to 1000 °C is close to that for the N720/A composite. The average α_t value for the N720/AM composite due to temperature rise from 23 to 1100 °C is slightly lower than that for the N720/A composite.

4.3 Monotonic Tension

To understand the tensile properties of N720/AM ceramic matrix composite, three specimens were subjected to tensile tests at 900, 1000 and 1100 °C. All tests were performed in laboratory air. Displacement control method at the constant rate of 0.05 mm/s was used. Tensile stress-strain curves for the N720/AM ceramic matrix composite at 900, 1000, and 1100 °C in air are illustrated in Figure 26.



Figure 26. Tensile stress-strain curves for N720/AM composite obtained at 0.05mm/s in the 900-1200 °C range.

For the specimens having $0^{\circ}/90^{\circ}$ fiber orientation, the fiber orientation is in the same direction of the load and dominates the mechanism in failure to tension. As a result, fiber volume fraction, V_f, must be considered as a critical parameter to facilitate comparison of the experimental results obtained from different composites. Previous work [18] for the N720/A ceramic matrix composite had a fiber volume fraction of 44%, but in this research the fiber volume fraction of the N720/AM ceramic matrix composite was 38.5%. After normalizing the UTS values to 44% V_f by using the relationship:

$$UTS_{adj} = (\frac{44}{38.5}) * UTS$$
, the adjusted UTS values for the N720/AM composite at 900,

1000 and 1100 °C become 167.3, 184 and 183 MPa, respectively.

The results of average tensile properties for N720/A and N720/AM ceramic

matrix composites are represented in Table 8.

Material	Elastic Modulus (GPa)	UTS (MPa)	Failure Strain (%)
Tests at 900 °C			
N720/A	70.1	190	0.33
N720/AM	77.1 ^a	167.3 ^a	0.32
<i>Tests at 1000 •C</i>			
N720/A	73.5	187.5	0.32
N720/AM	72.5 ^a	184 ^a	0.37
<i>Tests at 1100 •C</i>			
N720/A	69.6	186	0.33
N720/AM	72.5 ^a	183 ^a	0.4

Table 8. The average tensile properties for the N720/A and N720/AM composites atvarious temperatures. Data for N720/A from Braun[18,19]

^a Adjusted for V_f=0.44

As it is seen in Table 8, as the temperature increases the failure strain specimen increases for the N720/AM and it is noteworthy that the N720/A specimens have bigger UTS values at all temperature levels.

To identify the effects of loading rate on tensile properties of the N720/AM composite, four specimens were subjected to tension to failure at 1100 °C in steam. Tests were conducted with constant loading rates of 25 and 0.0025 MPa/s. The effect of loading rate on stress-strain curves of the N720/AM at 1100 °C in steam is shown in Figure 27.



Figure 27. Tensile stress vs. strain curves for the N720/AM with constant loading rates of 25 and 0.0025 MPa/s at 1100 °C in steam. Effect of loading rate on stress-strain behavior and strength properties is evident.

At the constant loading rate of 25 MPa/s, the tensile stress-strain curves seem nearly linear from beginning to failure. The average UTS obtained from two tests is 153 MPa (equivalent to 174.9 MPa for V_f = 0.44), the average elastic modulus was 63.8 GPa (equivalent to 72.9 GPa for V_f = 0.44), the average failure strain, 0.34%, is 0.06% lower than that measured in tensile test with displacement control in air. The UTS value is decreased by 4.5% and the modulus of elasticity is increased by 0.55% compare to those measured with displacement control at 0.05mm/s at 1100 °C in air.

At the constant loading rate of 0.0025MPa/s which is a change in loading rate by four orders, there is a significant change in the tensile stress-strain behavior of N720/AM composite at 1100 °C in steam. It can be seen in Figure 27 that the stress-strain curves

obtained at the constant loading rate of 0.0025 MPa are nonlinear to failure. The linearity of the both curves for 0.0025 MPa/s loading rate tests only remains until approximately 40 MPa stress level, then, a considerable deviation from linearity begins. The average UTS obtained from two tests at 1100 °C is 131 MPa. The average UTS value is much lower than that at the constant loading rate of 25 MPa/s. The average elastic modulus, 44.1 GPa, is 21% lower than that at 25 MPa/s. The failure strain values are ranging from 0.55 to 0.64% where the average is 0.6% which is 174%, of that obtained at 25 MPa/s. The average results for the N720/A and N720/AM ceramic matrix composites at 1100 and 1200 °C are presented in Table 9.

Table 9. The average tensile properties for N720/AM composite with constant loading rates at 25 and 0.0025MPa/s in steam. Data at 1200 °C from Genelin [16]

Temperature	Elastic	UTS	Failure	Time to				
(°C)	Modulus	(MPa)	Strain	Failure				
	(GPa)		(%)	(h)				
Loading rate at 25MPa/s								
1100	72,9 ^a	174.2 ^a	0.34	0.0017				
1200	50.1 ^a	149.6 ^a	0.6	14.57				
Loading rate at 00.25MPa/s								
1100	66.44 ^a	165.4 ^a	0.55	0.0016				
1200	27.64 ^a	103 ^a	1.26	10.38				
a A 1° / 1 C	V/ 0 44							

^a Adjusted for V_f=0.44

As it was reported for N720/A ceramic matrix composite at 1100 °C by Braun [18] and as it is seen in Table 9 for N720/AM ceramic matrix composite at 1100 and 1200 °C, a strong dependence of tensile behavior on loading rate was observed in this effort like in the previous researches.

4.4 Creep-Rupture

Creep tests were performed at 1100 °C in both air and steam and at 1000° C only in steam. Creep run-out time was defined as 100 hours in all tests. Results of creeprupture tests for N720/AM ceramic matrix composite are given in Table 10.

	Table 10. Results of Creep-Rupture tests for 11/20/AM.							
Specimen	Environment	Environment Temperature Creep Creep S						
(#)		(°C)	Stress	Strain	Life	Creep Rate		
			(MPa)	(%)	(h)	(s^{-1})		
6*	Air	1100	109	0.2	>100	2.29E-07		
7*	Air	1100	131	0.23	>100	3.32E-07		
8	Steam	1100	131	0.42	4.12	1.93E-05		
9	Steam	1100	109	0.87	35.2	5.81E-06		
10*	Steam	1100	87.5	0.69	>100	9.87E07		
11*	Steam	1000	131	0.20	>100	6.51E-08		
12*	Steam	1000	140	0.21	>100	2.17E07		

Table 10. Results of Creep-Rupture tests for N720/AM

*Run-out

Creep-rupture tests for N720/A ceramic matrix composite with 0°/90° fiber orientation were performed at 1100 °C in previous research by Braun [18]. He studied with N720/A specimens having fiber volume fraction of 44%. In contrast, in this effort fiber volume fraction was 38.5%. To facilitate the comparison of creep behavior of composites, equivalent creep stress levels must be used. Therefore; the creep stress values have to be adjusted to values for V_f=0.44 by using the relation: $\sigma_{adj} = (\frac{44}{38.5}) * \sigma$.

Consequently, instead of the creep stress levels 87.5, 109, 131 and 140 MPa, especially in the comparison charts, the equivalent creep stress levels of 100, 125, 150 and 160 MPa are used, respectively.

4.4.1 Creep-Rupture Tests at 1100°C

Two creep-rupture tests were performed at 1100 °C in air. At 109 MPa (equivalent to 125 MPa for V_f = 0.44), creep run out was achieved with a creep strain accumulation of 0.2%. Creep run-out was also achieved at 131 MPa (equivalent to 150 MPa for V_f = 0.44) in air. Accumulated creep strain was 0.23%.

As a result of increasing creep stress, creep strain accumulation increases as it was expected. It is worth to note that creep strain accumulations remain below < 0.25% at both stress levels. Although the creep strain accumulation is low for each stress levels, in previous effort [18] for the N720/A, creep strain accumulation was only 0.13% at 150MPa for V_f = 0.44. This is nearly 50% of that accumulated for the N720/AM in this effort. Note that, the N720/A specimen also achieved run-out at 150 MPa for V_f = 0.44 in air.

Notice that creep curves obtained at both 109 and 131 MPa stress levels show primary and secondary creep regimes. Figure 28 shows the creep strain vs. time curves at 1100 °C in air with the curves obtained in steam.

Three creep tests were performed at 1100 °C in steam at the creep stress levels of 131, 109 and 87.5MPa (equivalent to 100, 125 and 150 MPa for V_f = 0.44, respectively). Creep run-out stress was 87.5MPa. Results of the creep tests conducted at 1100 °C both for the N720/AM and N720/A composites are summarized in Table 11. Results for the N720/A composite from Braun [18] are included for comparison.

Material	Creep	Creep	Creep			
	Stress	Strain	Life			
	(MPa)	(%)	(h)			
Tests in laboratory air						
N720/A	150	0.13	>100			
	125 ^a	0.2	>100			
N720/AM	150 ^a	0.23	>100			
Tests in steam						
	100	0.50	>100			
N720/A	125	0.49	53.7			
	150	0.46	12.2			
	100 ^a	0.69	>100			
N720/AM	125 ^a	0.87	35.2			
	150 ^a	0.42	4.12			

Table 11. Results of Creep-Rupture tests for the N720/A and N720/AM composites with 0°/90° fiber orientation at 1100 °C. Data for N720/A from Braun[18]

^a Adjusted for V_f=0.44

Notice that the creep run-out stress level of 87.5MPa (equivalent to 100 MPa for $V_f = 0.44$) is same for both the N720/AM and N720/A tested in steam. Creep life times for the stresses of 109 MPa (equivalent to 125 MPa for $V_f = 0.44$) and 131 MPa (equivalent to 150 MPa for $V_f = 0.44$) for N720/AM are lower than those for N720/A. Figure 28 and 29 illustrate the creep strain vs. time curves for the N720/AM and N720/A composites at 1100 °C, respectively. It is noteworthy that curves for the same stress levels.

It can be seen in Figure 28 that the accelerated creep behavior in steam is apparent compared to the creep behavior in air at the same temperature. Furthermore; the acceleration rate increased with the applied creep stress. However, an increase in creep strain accumulation occurs as the stress level increases from 87.5 to 109 MPa and then a significant decrease occurs in the creep strain from 0.87% to 0.42% as the stress level

increase to 131 MPa. Similar behavior for the N720/AM ceramic matrix composite at 1200 °C was also reported by Genelin [16]. Creep strain-time curves at 1100 °C in steam also show primary and secondary creep regimes for all creep stresses in this research. Creep strain vs. time curves for 109 and 131 MPa creep stresses exhibit little tertiary regimes after showing primary and secondary regimes.



Figure 28. Creep strain vs. time curves for N720/AM composite at 1100 °C in air and steam. Stress levels adjusted for $V_f = 0.44$ are shown in parentheses.



Figure 29. Creep strain vs. time curves for N720/A composite at 1100 °C in air and steam. Data from reference [18].

4.4.2 Creep-Rupture Tests at 1000 °C

Two creep-rupture tests were performed at 1000 °C only in steam. Creep stress levels were 131MPa and 140MPa. At 131 MPa (equivalent to 150 MPa for $V_f = 0.44$), creep run out was achieved with a creep strain accumulation of 0.2%. Creep run-out was also achieved at 140 MPa (equivalent to 160 MPa for $V_f = 0.44$) in steam. Accumulated creep strain, 0.21%, is only 0.01% larger than that accumulated at 131MPa. Similar behavior for the N720/A ceramic matrix composite at 1000 °C was also reported by Braun [19]. Results of the creep tests conducted at 1000 °C both for the N720/AM and N720/A composites are summarized in Table 12. Results for the N720/A composite from reference [18] are included for comparison.

Material	Creep Stress (MPa)	Creep Strain (%)	Creep Life (h)		
Tests in laboratory air					
N720/A	150	0.08	>100		
Tests in steam					
	135	0.13	>100		
N720/A	150	0.14	>100		
	160	0.18	>100		
	150 ^a	0.2	>100		
N720/AM	160 ^a	0.21	>100		
	100	0.21	~100		

Table 12 Results of Creep-Rupture tests for the N720/A and N720/AM compositesat 1000 °C. Data for N720/A from Braun [18].

^a Adjusted for V_f=0.44

Creep strain accumulation increases as the applied creep stress increases, as it was expected. It is worth to note that creep strain accumulations remained very low at all stress levels. Although the creep strains produced for the N720/AM were < 0.22%, they are higher compare to the same stress levels for the N720/A composite. Figure 30 and 31 illustrates the creep strain vs. time curves for the N720/AM and N720/A composites at 1000 °C. Creep curves for the N720/A from Braun [18] are shown in Figure 31 for comparison.

Primary creep and secondary creep regimes are observed both for the N720/AM and the N720/A curves at all stress levels. Secondary creep until run-out is apparent for all stress levels conducted at 1000 °C. As it is seen in the Figure 30 and 31, the creep strain vs. time curves at the same stress levels of the N720/AM and N720/A exhibit similar trends at 1000 °C.



Figure 30. Creep strain vs. time curves for the N720/AM ceramic matrix composite at 1000 °C. Stress levels adjusted for $V_f = 0.44$ are shown in parentheses.



Figure 31. Creep strain vs. time curves for the N720/A ceramic matrix composite at 1000 °C. Data from reference [18]

4.5 Effects of Temperature and Steam on Creep Rupture Behavior

The effect of temperature on creep behavior for the N720/AM ceramic matrix composite in air at 1100 and 1200 °C can be seen in Figure 32. Creep curves at 1200 °C from Genelin's research [16] are included for comparison. At the adjusted stress levels of 125 and 150 MPa for $V_f = 0.44$ in air, creep strain accumulation increased significantly with the increase of temperature from 1100 to 1200 °C. Creep run-out was achieved at each stress level at 1000 °C, but in the case of 1200 °C creep life time decreased at each stress dramatically for the N720/AM composite.

As it is seen clearly in Figure 32b, the creep life time at the adjusted stress level of 150 MPa for $V_f = 0.44$ is only 0.6 hours. All of the creep strain vs. time curves exhibit primary and secondary creep regimes at each stress level. Secondary creep until run-out trend is apparent for all stress levels conducted at 1100 and 1200 °C in air.



Figure 32. Creep strain vs. time curves for N720/AM ceramic matrix composite at 1100 and 1200 °C in laboratory air. Time scale is reduced to clearly. show the creep curve at 150MPa. Data at 1200 °C from Genelin [16]. Stress levels adjusted for V_f = 0.44 are shown in parentheses.

Figure 33 shows the effect of temperature on creep behavior for the N720/AM ceramic matrix composite in steam at 1100 and 1200 °C. Creep curves at 1200 °C from Genelin [16] are also shown for comparison. At the adjusted stress level of 100 MPa for $V_f = 0.44$ in steam, increase in the amount of creep strain accumulation is obvious as the temperature increases from 1100 to 1200 °C, but the creep life time decreased significantly as it was expected. In contrast, accumulated strain decreases with the increase of temperature in the case of the adjusted creep stress of 125 MPa for $V_f = 0.44$. Consequently, increasing accumulated creep stress level.

As it was seen Figure 28, the effect of steam on creep behavior for N720/AM is apparent. The trend of curves reveals that the presence of steam has a significant effect on increasing the creep rates. Since presence of steam accelerates the creep rates, we have larger creep strains. This behavior was also reported by Braun [18], Mehrman [13] and Ruggles-Wrenn et al [24] for the N720/A composite at elevated temperatures. For the N720/AM composite creep strain accumulated at 131 MPa in steam is nearly two times the creep strain accumulated at the same stress level in air at 1100 °C. Furthermore; dramatically longer lifetimes were observed in air than in steam during creep tests at 1100 °C.

54



Figure 33. Creep strain vs. time curves for N720/AM ceramic matrix composite at 1000, 1100 and 1200 °C in steam. Data at 1200 °C from Genelin [16]. Stress levels adjusted for $V_f = 0.44$ are shown in parentheses.

Figure 34 illustrates the creep curves produced at a given applied stress level in air and steam to gain a further understanding of the effect of the environment on creep behavior for N720/AM at 1100 °C. Note that creep curves for N720/AM composite obtained in steam differ significantly from that obtained in air at 1100 °C for the creep stress level of 131 MPa (equivalent to 150 MPa for $V_f = 0.44$). Recall that creep run-out was achieved only at the equivalent stress of 100 MPa for $V_f = 0.44$ in steam at 1100 °C for N720/AM. Creep curves for N720/A at the creep stress levels of 150 MPa in both steam and air at 1100 °C from Braun [18] are also shown for comparison in Figure 34b.

It can be clearly seen that the behavior of the curves in Figure 34a and b are similar. In addition, larger creep strain was accumulated by the N720/AM than by the N720/A for the creep stress level equivalent to 150 MPa for $V_f = 0.44$ in air. In contrast, N720/A produced larger creep strain than N720/AM did in steam.



Figure 34. Creep strain vs. time curves for N720/A and N720/AM ceramic matrix composites at 150 MPa creep stress level at 1100 °C. Stress level adjusted for $V_f = 0.44$ is shown in parentheses. Data for N720/A from reference [18].

Minimum creep strain rate was reached in all tests. Creep rate as a function of applied stress for N720/AM and N720/A composites with 0°/90° fiber orientation in air and steam are illustrated in Figure 35 and 36. Notice that the secondary creep rates for the N720/AM are bigger than those for the N720A.



Figure 35. Minimum creep rate as a function of applied stress for N720/AM ceramic matrix composite at 1100 °C. Creep stress levels are adjusted to $V_f = 0.44$.



Figure 36. Minimum creep rate as a function of applied stress for N720/A ceramic matrix composite at 1100 °C. Data from reference [18].

Figure 37 summarizes the stress- rupture behavior of both the N720/A and N720/AM ceramic matrix composites where the results for the N720/A obtained from prior effort [18]. Notice that creep tests conducted at the equivalent stress levels of 150 and 160 MPa for $V_f = 0.44$ at 1000 °C in steam achieved run-out in this effort. Furthermore, the creep-rupture tests were performed in prior work [18] also achieved run-out at 150 MPa in air and 150 and 160 MPa in steam. As a result, increase in temperature from 1000 to 1100 °C has no effect on creep lifetime up to 100h, run-out time, for both N720/A and N720AM in steam and air. This behavior for N720/A was also reported by Braun [18]. Creep run-out stress for the N720/AM composite at 1100 °C was 150 MPa for V_f-0.44 in air. At 1200 °C, the effect of temperature was dramatically apparent in Genelin's work [16] where the the creep lifetime was only 0.59h for N720/AM at the stress level of 150 MPa for V_f=0.44 and the creep run-out stress was 100 MPa for V_f=0.44 in air. In steam at 1100° C the effect of temperature was also obvious. Run-out achieved at 100 MPa for V_f=0.44 and the creep lifetime was only 4.12h at 150 MPa for V_f=0.44 for N720/AM. In Braun's work [18] the creep-run out stress was also 100 MPa and the creep lifetime was 12.2h at 150 MPa.



Figure 37. Creep stress vs. time to rupture for N720/AM ceramic matrix composite at 1100 °C in air and steam. Data for N720/A from reference [18]. Stress levels of N720/AM are adjusted for $V_f = 0.44$.

4.6 Retained Tensile Properties

Retained strength and modulus of the specimens subjected to prior creep are presented in Table 13. The tensile stress-strain curves for the specimens subjected to prior creep at 1000 and 1100 °C are illustrated in Figure 38a and 38b, respectively. Stress-strain curves for the as processed material are also presented in figures for comparison.

Environment	Creep	Retained	Retained	Failure	Modulus	Strength
	Stress	Modulus	Strength	Strain	Retention	Retention
	(MPa)	(GPa)	(MPa)	(%)	(%)	(%)
Tests at 1000 •	С					
Steam	131	58.6	166.2	0.20	89.6	103.2
	140	60.6	173.2	0.21	92.8	107.6
Tests at 1100 •	С					
Air	109	59.8	168	0.2	91.7	104.9
	131	55.8	174.4	0.23	86.2	108.9
Steam	87.5	55.4	162.8	0.69	86	101.6

Prior creep in both air and steam had effect on tensile strength at 1000 and 1100 °C. For the specimens subjected to prior creep stress levels of 131 and 140 in steam at 1000 °C, tensile strength retentions were 103.7% and 107.6%, modulus losses were 10.4% and 7.2%, respectively. For the specimens subjected to prior creep stress levels of 109 and 131 in air at 1100 °C, the tensile strength increased by 4.9% and 8.9%, respectively. The minimum strength retention was observed for the specimen subjected to prior creep stress level of 87.5 in steam as 101.6%. However, the loss in modulus was 14%.



Figure 38. Effect of prior creep on tensile stress-strain behavior of the N720/AM composite at (a) 1000 °C (b) 1100 °C. Stress levels adjusted for $V_f = 0.44$ are shown in parentheses.

4.7 Microstructural Analysis

In this part of the chapter, the fracture surfaces of the N720/AM specimens are characterized. As it was mentioned in chapter 4, after the mechanical testing process, the specimens were examined using both an optical and a scanning electron microscope (SEM).

4.7.1 Optical Microscopy Analysis

Optical micrographs of the N720/AM specimens are presented in Figure 39 through 43. Possible effects of temperature, environment, creep stress level and lifetime are investigated.

The fracture surfaces obtained from the specimens subjected to tensile to failure at 900, 1000, 1100 and 1200 °C in air are presented in Figure 39. Specimen for 1200 °C obtained from prior work [16]. The similar damage zones can be seen for all temperature levels. The fibers in the 0° direction exhibit random fiber pull-out characterization. However, the damage zone appears relatively short at 900 °C and the specimens tested at 1000 and 1200 °C exhibit relatively longer damage zones. Consequently, a clear correlation between the fracture surface and the increasing temperature is not apparent.



Figure 39. Fracture surfaces of the N720/AM specimens tested in tension to failure in air at: (a) 900 (b) 1000 and (c) 1100 °C and (d) 1200 °C. Micrograph at 1200 °C was obtained from Genelin [22].

Figure 40 shows the fracture surfaces of the N720/AM specimens obtained in creep tests at 109 and 131 MPa in air at 1100 °C. Note that, the specimens achieved runout and then subjected to tension to failure. It can be seen that the size of the damage zone is large for each. The fiber pull-out lengths are clearly seen especially from the side views of micrographs.



Figure 40. Fracture surfaces of the N720/AM specimens obtained in creep tests at 1100 °C in air at: (a-b) 131MPa (equivalent to 150 MPa for V_f = 0.44), t_f = > 100 h and (c-d) 109 MPa (equivalent to 125 MPa for V_f = 0.44), t_f = > 100h.

Figure 41 illustrates the fracture surfaces of specimens obtained in the 87.5, 109 and 131 MPa creep tests in steam. The noticeable difference in the length of the damage zones is apparent. The damage zone for the specimens tested at 109 MPa is the longest among all. Since the specimen tested at 131MPa has the shortest damage zone, there is no correlation between creep stress level and damage zone length. Similar observation was reported by Genelin [16] for the N720/AM specimens at 1200 °C.



Figure 41. Fracture surfaces of the N720/AM specimens obtained in creep tests at 1100 °C in steam at: (a-b) 87.5MPa (equivalent to 100 MPa for V_f = 0.44), t_f = >100 h and (c-d) 109 MPa (equivalent to 125 MPa for V_f = 0.44), t_f = 35.2 h (e-f) 131 MPa (equivalent to 150 MPa for V_f = 0.44), t_f = 4.12.

Fracture surfaces of the N720/AM specimens tested in creep at the equivalent stress level of 150 MPa for V_f = 0.44 at 1000, 1100 and 1200 °C are presented in Figure 42. Correlation between the damage zone length and the increasing temperature is apparent for this case. Specimen for 1200 °C (obtained from Genelin [16] shows relatively short damage zone. As the temperature decreases from 1200°C to 1100 and 1000°C, the length of the damage zone increases significantly. The difference can be seen clearly from the side views of the specimens presented. (see Figure 42b, d and f) Note that the specimen tested at 1000 °C achieved run-out and subsequently subjected to tensile test to failure.



Figure 42. Fracture surfaces of the N720/AM specimens obtained in creep tests in steam at (a-b) 131 MPa (equivalent to 150 MPa for V_f = 0.44), t_f = > 100 h at 1000 °C and (c-d) 131 MPa (equivalent to 150 MPa for V_f = 0.44), t_f = 4.12 h at 1100°C (e-f) 136 MPa (equivalent to 150 MPa for V_f = 0.44), t_f = 0.012 h at 1200°C in steam. Micrographs of the specimen tested in 1200°C from Genelin [16].

To investigate the effect of loading rate on the fracture surface, optical micrographs of the N720/AM specimens subjected to tension to failure with constant loading rates of 25 and 0.0025 MPa are shown in Figure 43. From prior work at 1200° C, Genelin observed that the N720/AM specimen subjected to tensile to failure with the loading rate of 25 MPa has a relatively short damage zone and small amount of fiber pull-out [16]. In these figures below, there is a small difference between the damage zones, but the different is not so apparent compare to the figures of the same tests at 1200 °C obtained by Genelin [16]. The fracture surface of the specimen tested with the

constant load rate of 0.0025 MPa exhibit more brushier surface than that tested with the constant loading rate of 25 MPa/s.



Figure 43. Fracture surfaces of the N720/AM specimens subjected to tensile test to failure in steam at 1100 °C with the constant loading rates of: (a-b) 25 MPa/s, $t_f = 0.0017h$ and (c-d) 0.0025 MPa/s, $t_f = 10.38$ h.

4.7.2 Scanning Electron Microscopy Analysis

Optical micrographs presented in the previous section helped us to understand the general characteristic of the fracture surface. To obtain detailed information about the fracture surface, the SEM micrographs are inspected in this part of the chapter. Notice that scanning electron microscope allows us to get micrographs of the fracture surface of the specimens at higher magnifications.



Figure 44. Fracture surface of the N720/AM specimens tested in tension to failure in air at: (a-b) 900 °C (c-d) 1000 °C and (e-f) 1100 °C

The fracture surfaces of the specimens tested in tensile to failure at varies temperatures to understand the temperature effect on the fracture surface and to find a correlation between temperature and topography in Figure 44. The fracture surfaces contain large amounts of uncoordinated brushy 0° fibers especially at 900 and 1000 °C tests. The fracture surface at 1100 °C is dominated by planar failure. Although there are some differences in the brushiness and fibrousness, the three fracture surfaces have basically the same topography.

Figure 45 shows additional micrographs of the surface fracture for the N720/AM specimens tested in tensile to failure at various temperatures. Minimal 0° fiber pull-out which is typical for the specimens subjected to tensile can be seen in Figure 45a and b. Furthermore Figure 45c shows also 0° and 90° orientation after the failure and short fiber pull out also seen. Individual fiber with small amount matrix adhering on is shown in Figure 45d.



Figure 45. Fracture surfaces of specimens tested in tension to failure with displacement control at various magnifications in air at a) 1100 °C b) 1000 °C and (c-d) 900 °C

The fracture surface of two N720/AM specimens tested in creep at 109 and 131 MPa in steam at 1100 °C are showed in Figure 46. For the case of the specimen tested at 109 MPa, the fracture surface predominantly brushy with long fiber pull-out. Notice that the specimen at 109 MPa achieved run-out and subsequently subjected to tensile to

failure. In contrast, the fracture surface of the specimen at 131 MPa is dominated by planar areas with coordinated fiber failure.



Figure 46. Fracture surfaces of the N720/AM specimens obtained in creep tests at 1100 °C in steam at: (a-b) 131MPa (c-d) 109 MPa

Figure 47 shows the detailed fracture surfaces of the N720/AM specimens tested in creep at 131 and 109 MPa in air and in steam at 1100 °C. The effect of environment on the fracture surfaces is apparent. Figure 47b shows a greater amount of matrix adhering to the fibers.. Likewise, steam effect is also evident for the fibers of the N720/AM specimens at 109 MPa. Recall that the creep lifetime of the specimen at 109 MPa in steam at 1100 °C was 35.2 hours.



Figure 47. Detailed fracture surfaces of N720/AM specimens subjected to creep at 131 MPa at (a) 1100 °C in air (b) 1100 °C in steam and at 109 MPa at (a) 1100 °C in air and (b) 1100 °C in steam

Figure 48 shows the micrographs of the N720/AM specimens obtained at 131 MPa at 1000 and 1100 °C in steam. Matrix material remaining bonded to the fibers in the
specimen at 1000 and 1100 °C in steam is apparent in Figure 48b and d, respectively. Notice that the amount of matrix bonded to the fibers at 1000 °C is greater than that at 1100 °C in steam. Although the temperature level was higher, the creep life of the specimen at 1100 °C is only 4.12 hours that has the critical effect on the amount of matrix bonded to the fibers. Notice that the N720/AM specimen tested at 131 MPa at 1000 °C in steam achieved run-out of 100 hours.



Figure 48. Fracture surfaces of the N720/AM specimens subjected to creep at 131 MPa in steam at (a-b) 1000 °C and (c-d) 1100 °C.

Figure 49 shows the loading rate effect on the fracture surface of the N720/AM specimens with the micrographs at various magnifications. The fracture surface of the specimen tested at 0.0025 Mpa/s in Figure 49a and b exhibit more brushy uncoordinated fiber failure than that tested at 25 MPa/s in Figure 49d and f. The specimen tested in tensile to failure at the loading rate of 25 Mpa/s also has a similar topography like the

specimens tested in tensile to failure at displacement control shown in Figure 44. The effect of steam exposure time is also apparent in Figure 49c and f. As it was expected, larger amount of matrix bonded to the fibers at 0.0025 MPa test where time to failure is 14.5 hours.



Figure 49. Fracture surface of N720/A specimen subjected tensile to failure at 1100 °C in steam with the constant loading rates of: (a-b-c) 25MPa/s and (d-e-f) 0.0025 MPa/s

V. Conclusions

Tensile-strain behavior was investigated and tensile properties measured at 900, 1000 and 1100 °C. The UTS values for N720/AM are below the corresponding values for N720/A at 900, 1000, and 1100 [18,19]. The average elastic modulus, 65.5GPa for N720/AM (equivalent to 74,9GPa for V_f = 0.44), was higher than for N720/A [18,19]. Similar tensile comparison results were reported for the N720/AM and the N720/A composites at 1200 °C [16,22].

The influence of loading rate on tensile behavior also investigated with constant loading rates of 25 and 0.0025 MPa at 1100 °C in steam. At the constant loading rate of 0.0025 MPa/s, which is a change in loading rate by four orders, a significant decrease in the tensile stress-strain behavior of the N720/AM was observed. The failure strain values ranges from 0.55 to 0.64% where the average is 174% of that obtained at 25 MPa/s. The average tensile modulus and strength were 21% and 15% lower than those at 25 MPa/s, respectively. As it was reported for N720/A composite at 1100 °C by Braun [18] and for N720/AM composite at 1100 and 1200°C [ref. 16], a strong dependence of tensile behavior on loading rate was observed in this effort.

Creep tests were conducted at 1000 and 1100 °C in air and steam. At 1000 °C, tests were conducted at 131 and 140 MPa in steam. Tests at both 131 and 140 MPa achieved run-out. Creep strain accumulations were 0.2 and 0.21%, respectively. Creep run-out stress was 140 Mpa in steam at 1000 °C

At 1100 °C in air creep run out achieved at 109 and 131 MPa. Accumulated creep accumulations were 0.2 and 0.23%, respectively. Creep stresses were 87.5, 109 and 131

MPa in steam. Creep run-out stress was 87.5 MPa in steam at 1100 °C. Strain accumulations in steam at 1100°C were larger than those at 1000 °C. Largest creep strain 0.87% was observed at 109 MPa in steam at 1100 °C. In air at 1100 °C, the largest creep strain accumulated was 0.23%.

140 MPa, creep run-out stress in steam at 1000 °C, was 87% of UTS and 87.5 MPa, creep run-out stress in steam at 1100 °C, was only 54.7% of UTS. At all creep stresses, the strain accumulation of the N720/AM was higher than that of the N720/A [18] except at 131 MPa (equivalent to 150 MPa for $V_f = 0.44$) in steam at 1100 °C.

In creep-rupture tests, creep strain accumulation increased with the applied creep stress and/or temperature as it was expected. It is worth to note that creep strain accumulations remained very low especially at 1000 °C in steam and 1100 °C in air. Although the creep strains produced for the N720/AM were very low, they are higher compared to the same stress levels for the N720/A composite [18]

At 1000 °C in steam and 1100 °C in air the creep run out stresses of the N720/AM composite were same as those of the N720/A composite, but on the other hand the creep lifetimes of the failed N720/AM specimens were lower than those of the N720/A specimens.

Minimum creep rate reached in all tests. At 1100 °C, creep strain rates ranged from 2.2 x 10^{-7} to 3.3 x 10^{-7} in air and from 9.8 x 10^{-7} to 1.9 x 10^{-5} in steam. At 1000 °C, strain rates were 6.5 x 10^{-8} at 131 MPa and 2.17 x 10^{-7} at 140 MPa in steam.

Specimens achieved run-out were subjected to tensile tests to failure in order to find the retained properties. All specimens retained at least 100% of their tensile strength.

At 1000 °C, tensile strength retentions were 103.7 and 107.6, modulus losses were 10.4% and 7.2% at 131 and 140 MPa, respectively. At 1100 °C, the tensile strength increased by 4.9% and 8.9% in air at 109 and 131 MPa, respectively. The minimum strength retention was observed for the specimen subjected to prior creep stress level of 87.5 in steam as 101.6%. However, the loss in modulus was 14%.

Appendix. Additional Micrographs



Figure 50. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 51. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 52. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 53. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 54. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 55. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 56. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 57. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 58. . Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 59. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 60. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 61. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 62. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 63. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 64. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 65. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 66. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 67. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 68. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 69. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 70. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air



Figure 71. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 72. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 73. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air.



Figure 74. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 75. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 76. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 77. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 78. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 79. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 80. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 81. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 82. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 83. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 84. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 85. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 86. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 87. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 88. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 89. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 90. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 91. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 92. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 93. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 94. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 95. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 96. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 97. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 98. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 99. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 100. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 101. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 102. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 103. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 104. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 105. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 106. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 107. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air


Figure 108. . Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 109. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 110. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 111. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 112. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 113. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 114. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 115. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 116. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 117. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 118. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 119. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 120. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air.



Figure 121. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air.



Figure 122. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air.



Figure 123. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air.



Figure 124. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air.



Figure 125. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air.



Figure 126. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 127. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 128. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 129. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 130. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 131. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 132. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 133. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air



Figure 134. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air



Figure 135. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 136. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 137. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 138. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 139. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 140. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 141. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 142. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 143. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 144. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 145. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 146. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 147. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 148. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 149. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 150. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 151. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 152. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 153. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 154. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 155. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 156. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 157. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 158. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 159. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 160. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 161. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 162. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 163. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 164. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 165. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 166. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 167. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 168. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 169. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 170. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 171. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 172. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 173. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 174. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 175. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 176. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 177. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 178. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 179. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.


Figure 180. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 181. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 182. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 183. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 184. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 185. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 186. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 187. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 188. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 189 Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 190 Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 191 Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 192 Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 193 Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 194 Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 195 Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 203. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 196. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 197. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 198. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 199. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 200. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 201. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 202. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 203. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 204. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 205. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 206. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 207. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 208. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 209. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 210. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 211. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 212. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 213. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 214. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 215. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 216. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 217. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 218. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 219. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 220. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 221. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 222. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 223. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 224. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 225. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 226. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 227. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 228. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 229. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 230. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 231. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 232. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 233. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 234. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 235. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 236. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 237. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 238. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 239. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 240. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 241. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 242. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 243. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 244. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 245. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 246. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 247. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 248. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 249. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 250. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.


Figure 251. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 252. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 262. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C



Figure 253. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 254. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 255. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 256. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 257. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 258. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 259. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 260. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 261. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 262. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 263. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 264. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 265. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 266. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 267. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 268. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 269. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 270. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 271. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 272. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 273. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 274. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 275. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 276. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 277. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 278. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 279. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 280. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 281. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 282. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 283. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 284. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 285. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 286. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 287. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 288. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 289. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 290. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 291. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 292. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



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Figure 294. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 295. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 296. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 297. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 298. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 299. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 300. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 301. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 302. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 303. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 304. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 305. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 306. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 307. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



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Figure 309. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 310. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 311. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 312. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 313. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 314. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 315. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 316. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 317. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 318. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 319. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 320. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.


Figure 321. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



Figure 322. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



Figure 323. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



Figure 324. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C..



Figure 325. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



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Figure 332. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



Figure 333. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



Figure 334. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



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Figure 338. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



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Figure 340. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



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Figure 342. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



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Figure 344. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



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Figure 350. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



Figure 351. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



Figure 352. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 353. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 354. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 355. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 356. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 357. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 358. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



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Figure 360. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 361. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 362. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 363. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 364. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 365. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



. Figure 366. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 367. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 368. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 369. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 370. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 371. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



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Figure 373. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



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Figure 375. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 376. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 377. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



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Figure 382. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 383. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 384. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 385. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 386. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 387. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 388. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 389. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 390. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 391. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 392. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.


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Figure 394. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 395. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 396. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 397. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 398. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 399. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 400. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 401. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



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Figure 406. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



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Figure 408. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 409. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



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Figure 411. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 412. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



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Figure 414. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



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Figure 451. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 452. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 453. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 454. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 455. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 456. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



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Figure 459. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 460. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 461. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 462. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 463. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



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Figure 466. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



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Figure 474. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 475. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 476. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 477. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 478. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



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Figure 480. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 481. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 482. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 483. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 484. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 485. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 486. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 487. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 488. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 489. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 490. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 491. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 492. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 493. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 494. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 495. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 496. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 497. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 498. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 499. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 500. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 501. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 502. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 503. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air.



Figure 504. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air.



Figure 505 Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air.



Figure 506. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air.



Figure 507. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air (side view).



Figure 508. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air (side view).



Figure 509. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air (side view).



Figure 510. Fracture surface of the N720/AM specimen tested in tension to failure at 900°C in laboratory air (side view).



Figure 511. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air.



Figure 512. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air.



Figure 513. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air.



Figure 514. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air.



Figure 515. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air (side view).



Figure 516. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air (side view).



Figure 517. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air (side view).



Figure 518. Fracture surface of the N720/AM specimen tested in tension to failure at 1000°C in laboratory air (side view).



Figure 519. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air.



Figure 520. Fracture surface of the N720/AM specimen tested in tension to failure at 1100° C in laboratory air.



Figure 521. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air.



Figure 522. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air.



Figure 523. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air (side view).



Figure 524. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air (side view).



Figure 525. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air (side view).



Figure 526. Fracture surface of the N720/AM specimen tested in tension to failure at 1100°C in laboratory air (side view).



Figure 527. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 528. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 529. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 530. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 531. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam (side view).



Figure 532. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam (side view).



Figure 533. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam (side view).



Figure 534. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam (side view).



Figure 535. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 536. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 537. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 538. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam.



Figure 539. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam (side view).



Figure 540. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam (side view).



Figure 541. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam (side view).



Figure 542. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam (side view).



Figure 543. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam (side view).



Figure 544. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 25MPa at 1100°C in steam (side view).



Figure 545. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 546. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.


Figure 547. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam (side view).



Figure 548. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam (side view).



Figure 549. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam (side view).



Figure 550. Fracture surface of the N720/AM specimen tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam (side view).



Figure 551. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 552. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 553. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 554. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam.



Figure 555. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam (side view).



Figure 556. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam (side view).



Figure 557. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam (side view).



Figure 558. Fracture surface of the N720/AM specimen(2) tested in tension to failure with constant loading rate of 0.0025MPa at 1100°C in steam (side view).



Figure 559. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 560. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 561. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 562. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C.



Figure 563. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C (side view).



Figure 564. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C (side view).



Figure 565. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C (side view).



Figure 566. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in laboratory air at 1100°C (side view).



Figure 567. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 568. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 569. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 570. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C.



Figure 571. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C (side view).



Figure 572. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C (side view).



Figure 573. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C (side view).



Figure 574. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in laboratory air at 1100°C (side view).



Figure 575. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 576. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 577. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 578. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C.



Figure 579. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C (side view).



Figure 580. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C (side view).



Figure 581. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C (side view).



Figure 582. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1100°C (side view).



Figure 583. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



Figure 584. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



Figure 585. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



Figure 586. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C.



Figure 587. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C (side view).



Figure 588. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C (side view).



Figure 589. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C (side view).



Figure 590. Fracture surface of the N720/AM specimen tested in creep at 109 MPa in steam at 1100°C (side view).



Figure 591. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 592. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 593. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 594. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C.



Figure 595. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C (side view).



Figure 596. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C (side view).



Figure 597. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C (side view).



Figure 598. Fracture surface of the N720/AM specimen tested in creep at 87.5 MPa in steam at 1100°C (side view).



Figure 599. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 600. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 601. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C.



Figure 602. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C. 346



Figure 603. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C (side view).



Figure 604. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C (side view).



Figure 605. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C (side view).



Figure 606. Fracture surface of the N720/AM specimen tested in creep at 131 MPa in steam at 1000°C (side view).



Figure 607. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 608. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 609. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 610. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C.



Figure 611. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C (side view).



Figure 612. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C (side view).



Figure 613. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C (side view).



Figure 614. Fracture surface of the N720/AM specimen tested in creep at 140 MPa in steam at 1000°C (side view).

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Vita

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