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10-layer woven	Kevlar armo	r held in a squa	re steel frame and imp	pacted at norm	al incider	nce by a 9 mm FMJ (full metal jacket), 124
grain projectile.	. The compos	ite armor is dis	cretized into weft and	warp yarns to	simulate	its woven structure. The yarn is modeled as
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# Effect of frame size, frame type, and clamping pressure on the ballistic performance of soft body armor

G.M. Zhang <sup>a</sup>, R.C. Batra <sup>a,\*</sup>, J. Zheng <sup>b</sup>

<sup>a</sup> Department of Engineering Science and Mechanics, M/C 0219, Virginia Polytechnic Institute and State University, Blacksburg, VA 24061, United States <sup>b</sup> Program Executive Office – Soldier, US Army, 15395 John Marshall Highway, Haymarket, VA 20169, United States

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In memory of Professor Kevin Granata who was killed on 16 April 2007 during the massacre in Norris Hall, Virginia Tech.

#### Abstract

We analyze, with the computer code LS-DYNA, three-dimensional (3D) transient deformations of a 10-layer woven Kevlar armor held in a square steel frame and impacted at normal incidence by a 9 mm FMJ (full metal jacket), 124 grain projectile. The composite armor is discretized into weft and warp yarns to simulate its woven structure. The yarn is modeled as a 3D continuum. We consider failure of the yarn, and friction between adjoining layers and between the armor and the frame bars. For the armor perfectly bonded to the rigid frame bars, the computed residual speed and the residual kinetic energy of the projectile are found to increase with a decrease in the frame size implying thereby that the armor fixed in a smaller frame will have lower  $V_{50}$  than that of the same armor clamped in a larger frame. (The  $V_{50}$  of an armor equals the speed of a standard projectile that upon normal impact has 50% probability of just perforating the armor). For the armor allowed to slide between the frame bars, we have studied the effect of the pressure applied to the bars of the two- and the four-bar frames on the speed and the kinetic energy of the residual projectile. For both the two- and the four-bar frames, the speed of the residual projectile is found to increase with an increase in the applied pressure. Computed results also show that the armor fixed in the two-bar frame exhibits higher impact resistance than that held in the four-bar frame. The  $V_{50}$  is found to be ~270 m/s when the woven armor is held in a four-bar frame with a clamping pressure of 200 MPa. The  $V_{50}$  decreases with an increase in the pressure applied to either the two-bar or the four-bar frames. © 2007 Elsevier Ltd. All rights reserved.

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#### 1. Introduction

Composite materials have been widely used in many high-performance structures such as protective clothing, bullet-proof vests and helmets due to their high-specific strength and stiffness. The ballistic performance of soft body armor is characterized by  $V_{50}$ , which is usually determined experimentally, and equals the velocity of the projectile that upon normal impact on the armor has 50% probability of penetrating it.

Parameters affecting the ballistic performance of composite armor include material properties of the yarn, woven structure of the armor, projectile geometry, projectile velocity and its material, boundary conditions imposed on the armor, friction between the yarns, and friction between the yarn and the projectile. Duan et al. [1] used LS-DYNA to delineate effects of frictional forces on the ballistic performance of one-layer woven rectangular composite with all four edges either clamped or only two opposite edges clamped. However, they did not consider the failure of the projectile and the composite. A recent review paper [2] has discussed the effect of different material and geometric parameters on the ballistic performance of soft body armor.

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Corresponding author. Tel.: +1 540 231 6051; fax: +1 540 231 4574. *E-mail address:* rbatra@vt.edu (R.C. Batra).

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Generally in ballistic experiments the boundary of the armor system is held in a rectangular frame with pressure applied to the frame bars to hold the armor in place. Two different frames, namely, two-bar and four-bar, are employed. Also frame size can be varied by adjusting the distance between the opposite bars of the frame. Shockey et al. [3] experimentally ascertained the effect of boundary conditions on the ballistic performance of the armor and found that for both the 25 g blunt and the 26 g sharp fragment simulating projectile (FSP), the armor fixed on two opposite edges rather than on all four edges was more effective in reducing the kinetic energy of the projectile. Since experiments are very expensive to perform, it will be more economical if one could accurately delineate computationally the effect of the frame size and the pressure applied to its bars on the  $V_{50}$ . We note that small values of the applied pressure may not hold the armor well, and when impacted it will slide between the frame bars. However, very large values of this pressure may fracture the armor within the frame bars. Thus the ballistic performance of the armor is likely to depend upon the pressure applied to the frame bars and the frictional force between the yarn and the frame bars. Lee et al. [4] have studied experimentally the effect of the clamping pressure on the penetration resistance of a 5-ply composite laminate and found that the loss of the kinetic energy of the projectile decreased with an increase in the clamping pressure.

Hundreds of parallel high-strength and high-modulus fibers are grouped together to form a yarn and yarns are woven to form a single-ply fabric. It is still not possible to consider each fiber individually because of enormous computational resources required. A possibility is to model woven armor as an assembly of one-dimensional (1D) bar elements [5,6]. Tan and Ching [7] replaced the one-layer composite with a network of viscoelastic bars. For suitable values of material parameters, they found that computed results agreed very well with the ballistic test data. Barauskas and Abraitiene [8] simulated the armor with thin shell elements of thickness equal to that of the yarn. A more realistic discretization of the composite is obtained by using 3D solid elements that can account for orthotropic material properties, inter-yarn and inter-layer friction, material failure and undulations in the woven yarns. Gu [9] considered the actual structure of plain-woven fabrics and developed 3D finite element discretization of the woven composite into weft and warp yarns. The multi-layered woven composite was impacted by a steel projectile and the computed results were compared with the experimental data. However, the failure of the projectile was not considered.

There are three methods to determine the ballistic limit of a soft armor. An accurate but very expensive method is to carry out a large number of ballistic experiments. However, it is tedious to experimentally characterize the effect on  $V_{50}$  of each parameter, such as the projectile shape and material, armor material, armor thickness, and armor architecture. An alternative is to employ an

approximate model [10] of the armor system, analyze the problem analytically and establish scaling laws. The success in this case depends upon our understanding of the mechanisms involved in the penetration process and how well they can be incorporated in the analytical model. The third possibility is to use a numerical method such as the finite element method that finds an approximate solution of the pertinent initial-boundary-value problem but can incorporate realistic material behavior, complex geometries, friction effects, and material failure. The analysis can be easily modified when additional information on the material response and failure becomes available. After the mathematical model and the computational algorithm have been validated one can perform parametric studies, determine the V<sub>50</sub>, and also delineate parameters to which it is most sensitive. In this case V<sub>50</sub> equals the minimum projectile velocity with which the target when impacted at normal incidence is penetrated completely. A few experiments are needed to validate this technique.

Sun and Potti [11] proposed the following relation

$$E_{\rm DP} = \frac{1}{2}m(V_{\rm s}^2 - V_{\rm R}^2)$$

among the initial velocity  $V_{\rm s}$ , the residual velocity  $V_{\rm R}$  of the projectile of mass *m*, and the energy  $E_{\rm DP}$  required to completely perforate a target. Here  $E_{\rm DP}$  is assumed to be constant, and the projectile not to fail during the penetration process. This relation does not account for the energy required to deform the armor, and that dissipated due to friction effects. Lim et al.'s [12] simulation of ballistic impact of fabric armor with LS-DYNA showed that the energy absorbed during the penetration process increased with an increase in the incident speed when it is between the  $V_{50}$  and a critical value. For an initial speed greater than the critical value, the energy absorbed decreased suddenly. Zeng et al.'s [13] simulations of ballistic impact of woven fabric armor gave similar results.

Here we have used the commercial software LS-DYNA to numerically simulate 3D deformations of a woven Kevlar armor held in a rectangular frame and impacted at normal incidence by a hemispherical nosed cylindrical lead projectile coated with a thin layer of copper with the goal of finding the effect on the  $V_{50}$  of the frame size, the clamping pressure applied to the frame bars, and whether the frame has four-bars or only two opposite bars. We account for the failure of the projectile and the target during the penetration process, simulate the relative movement between the adjacent yarns, assume the Kevlar armor to be an orthotropic material, regard each layer of the woven composite as made of weft and warp yarns, and divide each yarn into 3D solid elements. It is found that the frame type and the pressure applied to its bars influence the ballistic performance of the armor and its  $V_{50}$ . This information should be useful to armor designers, and to those involved in certifying acceptable armor performance.

The rest of the paper is organized as follows. Section 2 describes the material and the geometric parameters of the armor and the projectile, constitutive relations and failure criteria used in LS-DYNA, and values assigned to different parameters. Results are described in Section 3, where effects of different material and geometric parameters on deformations of the armor and the projectile are also delineated.

#### 2. Material and geometric parameters

The hemispherical nosed projectile is the Remington 9 mm full metal jacket (FMJ), 124 grain (8.0 g), 13.3 mm long, comprised of 0.5 mm thick outer copper layer coated on the inner solid lead part. The bullet, its section through the centroidal axis, and their discretizations into 8-node brick elements are exhibited in Fig. 1. The total number of nodes and elements equal 37,885 and 35,376, respectively. We use the Johnson-Cook (JC) relation to simulate the thermoviscoplastic response of copper, and model lead as an elastic perfectly plastic material; each material is assumed to be isotropic. We also use the JC relation to compute damage induced in copper, and have listed in Tables 1–3 values assigned to material parameters. A material point of copper is taken to have failed when the damage parameter for it equals 1.0.

The woven Kevlar armor comprised of 28 uniform 0.25 mm thick layers is modeled as an orthotropic material. Even though the woven composite armor is made of yarns and each yarn is made of fibers, we could not consider each fiber individually, because of the enormous computational resources required. Instead, each yarn is considered as a continuum; a typical yarn and its discretization into 8-node brick elements is shown in Fig. 2a where the sine-wave shape of the yarn has been approximated by a rectangular-wave. Orthogonal yarns constitute one layer depicted in Fig. 2b of the armor. The yarns along the x- and the y-directions are called warp and weft, respectively. In our simulations the yarns at crossovers have an initial gap of

Table 1

Values of material parameters for copper in the JC thermoviscoplastic relation

Parameter		Value
A (GPa)	:	0.09
B (GPa)		0.292
С		0.025
n		0.31
m		1.09
Density $(kg/m^3)$		8950
Specific heat (J/kg K)		385
Shear modulus (GPa)		47.27
Bulk modulus (GPa)		102.4

Table 2

Values of material parameters for copper in the JC damage relation

Parameter	Value
	1.0
$D_2$	0
$D_3$	0
$D_4$	0
$D_5$	0
$T_{\rm m}$ (K)	1356
$\sigma_{\text{spall}}$ (GPa)	1.9

Values of material parameters for lead

Mass density	Young's	Poisson's	Yield stress	Failure
(kg/m <sup>3</sup> )	modulus (GPa)	ratio	(GPa)	strain
11,340	16	0.44	0.383	0.3

0.01 mm between them. The length and the width of a horizontal element equal 0.75 mm, and the projection of an oblique element on a horizontal plane equals 0.25 mm. In order to reduce computer memory requirements, we replaced the 28 uniform 0.25 mm thick layers by 10 uniform 0.70 mm thick layers.

We used the \*Mat\_composite\_damage model in LS-DYNA [14] to simulate the mechanical response of



Fig. 1. The discretization of the projectile/bullet into finite elements.



Fig. 2. The discretization into finite elements of (a) one yarn, and (b) one woven layer.

composite yarns with following values assigned to different material parameters:

# $\rho = 1440 \text{ kg/m}^3$ , $E_a = 164.00 \text{ GPa}$ , $E_b = E_c = 3.28 \text{ GPa}$ , $v_{ba} = v_{ca} = v_{cb} = 0.0$ , $G_{ab} = G_{bc} = G_{ca} = 3.28 \text{ GPa}$ ,

Shear strength in the ab plane = 1.886 GPa;

Longitudinal tensile strength along a-axis = 2.886 GPa; Transverse tensile strength along b-axis = 1.486 GPa; Transverse compressive strength along b-axis = 1.7 GPa;

Normal tensile strength along c-axis = 1.486 GPa;

Transverse shear strength in ca-plane = 1.586 GPa;

Transverse shear strength in cb-plane = 1.886 GPa.

Here a-axis is aligned along the direction of the yarn, the b-axis is the transverse direction in the plane of the layer, and the c-axis is along the normal to the ab-plane.

We note that in the "Mat\_composite\_damage model failed elements are not deleted from the computation. Thus severe distortions due to large deformations of even one element will drastically reduce the time step size needed to find a stable solution of the governing equations that will either stop computations completely or make them progress extremely slowly. This is overcome by also using the failure model "Mat\_add\_erosion, regarding the material in an element to have failed when the maximum principal strain at its centroid equals 0.2, and deleting the failed element from the analysis.

A small 0.01 mm gap is initially assumed between two adjoining layers of the 10-layer composite with the "Contact\_automatic\_surface\_to\_surface algorithm employed to simulate contact between them and prevent their interpenetration during the deformation process. The Coulomb friction force between adjacent layers, between adjoining yarns, and between the composite armor and the frame, is modeled by taking the coefficient of friction to be 0.3. The coefficient of Coulomb friction between the projectile and the composite armor is also set equal to 0.3.

#### 3. Results

#### 3.1. Effect of frame size

In ballistic experiments designed to find the V<sub>50</sub> of an armor, the armor is often held in a  $2a \times 2a$  steel frame with flat bars of width *h* that are pressed together with a pressure *P* applied to the bars; a typical frame is shown in Fig. 3. We ascertain the effect of the frame size on deformations of the armor by finding the residual velocity of the projectile moving at 400 m/s and impacting at normal incidence the armor held in the frame with 2a equal to 30.75 mm, 40.75, 50.75, 60.75, 70.75 and 80.75 mm, and a/r = 3.42, 4.53, 5.64, 6.75, 7.86 and 8.97, respectively. Here *r* equals the radius of the bullet. One expects that the effect of boundary conditions on deformations of the armor will diminish with an increase in the value of a/r.



Fig. 3. The sketch of a typical square frame.



Fig. 4. Discretization of the  $30.75 \text{ mm} \times 30.75 \text{ mm}$  composite armor into finite elements.

#### 3.1.1. Armor perfectly bonded to a rigid frame

In the first set of simulations, we regard frame bars as rigid, stationary and perfectly bonded to the armor. For the 30.75 mm  $\times$  30.75 mm armor, Fig. 4 shows the discretization of the woven armor into finite elements with one element along the thickness of each yarn. The number of elements increases quadratically with an increase in the value of 2*a*, and is listed in Table 4 for the above-stated six values of 2*a*. For 2a = 80.75 mm, the number of nodes exceeds one million with over 3 million degrees of freedom. Accordingly, larger frames are not considered here.

For 2a = 40.75, 60.75 and 80.75 mm, Fig. 5 exhibits deformed shapes of the armor and the bullet at t = 50and 95 µs. Results for every other frame size are exhibited in order to reduce the length of the paper. At t = 50 µs, the frame size has very little effect on deformations of the projectile and the armor, and the material near peripheries of the larger-size frame stays essentially undeformed. However, at t = 95 µs, deformations in the armor have propagated to the frame edges for 2a = 40.75 and 60.75 mm, but for 2a = 80.75 mm, a small portion of the armor near the frame edges has undergone very little deformations signifying that the frame size considered is sufficient. Whereas

Table 4

Number of	nodes and	elements	for differe	nt frame s	izes	
Size 2a (mm)	30.75	40.75	50.75	60.75	70.75	80.75
Nodes Elements	153,760 37,820	268,960 66,420	416,160 103,020	595,360 147,620	806,560 200,220	1,049,760 260,820

the bullet has perforated the armor for 2a = 40.75 and 60.75 mm, it is still piercing the armor for 2a = 80.75 mm signifying that the tail-end velocity depends upon the frame size. For smaller size frames and thus armor, there is less armor material to absorb the kinetic energy of the projectile, and a larger volume fraction of the armor material enclosed in the frame is severely deformed and fails. For the same lateral deflection of the armor in front of the bullet nose larger axial strains are induced in yarns of the small size frame than those in the large size frame.

Time histories of the speed of the tail-end of the projectile for different frame sizes are exhibited in Fig. 6. When the elastic wave induced by the impact of the projectile with the armor reaches the tail-end of the projectile the speed of the tail-end begins to drop. Up to 50 µs after impact, there is not much difference among the tail-end velocities for different frame sizes. It is obvious that the speed of the residual projectile increases with a decrease in the frame size signifying that smaller size armors either cause less of the bullet material to fail or the armor in front of the bullet quickly fails thereby reducing resistance offered to the bullet. Oscillations occur in the velocity of the tailend of the projectile due to the back and forth propagation of the stress wave in the projectile. For the largest size armor considered here, the tail-end speed decreases affinely with time for  $t > 32 \,\mu s$ . Results documented in Fig. 6 confirm those included in Fig. 5 in that the lateral dimensions or the size of the armor strongly influence its deformations and hence the computed  $V_{50}$  of the projectile:

For the six frame sizes considered, Fig. 7a and b exhibit time histories of the projectile kinetic energy and its



Fig. 5. Deformed shapes of the armor and the projectile at (a) 50 µs and (b) 95 µs for three different frame sizes (from left to right) 40.75 mm, 60.75 mm and 80.75 mm.



Fig. 6. Time histories of the speed of the tail-end of the projectile for different frame sizes.

residual mass. For  $t > 50 \ \mu s$  both the projectile mass and the projectile kinetic energy decrease with an increase in the frame size. Assuming that projectile's average speed is nearly the same as that of its tail-end, the projectile kinetic energy decreases with an increase in the frame size due to both a decrease in the projectile mass and its speed. That is, the volume of the failed projectile increases with an increase in the frame size. Note that there are no oscillations in these curves, and the kinetic energy of the residual projectile for the 50.75 mm frame is greater than that for the 60.75 mm frame.

Fig. 8 depicts the relation between the kinetic energy of the projectile that is used up during the penetration process and the frame size. The horizontal solid line represents the initial kinetic energy of the projectile. Nearly 62.5% of the kinetic energy is dissipated during the penetration process for the 30.75 mm frame and this number increases to 92.6% for the 80.75 mm frame. Thus the frame size noticeably affects the kinetic energy of the residual projectile.

# 3.2. Armor held by uniform pressure applied to the four-bar frame

As mentioned above, in ballistic experiments, the armor is usually held between the frame bars by uniform pressure applied to them. The applied pressure should be below the compressive strength of the bar material and of the armor, otherwise one of these two will fail prior to the start of the test. Upon impact of the bullet with the armor, the relative movement of the armor between the frame bars will depend upon the applied pressure and the coefficient of friction between the armor and the material of the frame bars. Since steel used for the frame bars has much higher Young's modulus and compressive strength than the yarn, it is reasonable to regard the frame bars as rigid. We now investigate the effect of the relative sliding of the armor between the frame bars on armor's deformations.

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Fig. 7. For different frame sizes, time histories of the (a) kinetic energy, and (b) mass of the projectile.

A typical system comprised of a four-bar steel frame, the armor and the bullet employed during ballistic tests is shown in Fig. 9. Each 42 mm long, 6 mm wide and 0.5 mm thick frame bar is divided into uniform 0.5 mm  $\times$  0.5 mm solid elements. A uniform clamping pressure P = 10, 25, 100, 200 or 300 MPa is applied to the frame bars. Note that the frictional force between the frame bars and the armor will not necessarily be uniformly distributed since deformations of the armor between the bars may be inhomogeneous. The finite element mesh in the frame bars will help simulate this variation in the frictional force.

Deformed configurations at t = 20, 50, 80 and 95 µs and for P = 10, 25, 100 and 200 MPa (results for P = 300 MPa are not depicted since they are very similar to those for P = 200 MPa) of the projectile and the armor are shown in Fig. 10 for initial bullet speed of 400 m/s. For each one of the four values of the pressure, deformations of



Fig. 8. The variation with the frame size of the reduction in the kinetic energy of the projectile.



Fig. 9. Schematic sketch of the system comprised of the armor, the fourbar frame, and the bullet.

the system are essentially identical at 20  $\mu$ s, but are different at later times in the following two respects. First, the thickness of the composite armor ahead of the bullet is different. With an increase in the pressure applied to the frame bars, more of the armor ahead of the bullet fails. At  $t = 80 \mu$ s, the bullet has not perforated the armor for P = 10 MPa, but the bullet nose has come out of the armor for P = 100 and 200 MPa. The second difference is that for smaller values of the clamping pressure P more of the armor material moves towards the center of the frame; this becomes transparent for  $t > 50 \mu$ s. This inward motion of the armor facilitates the transverse displacement of the composite ahead of the bullet. At  $t = 95 \mu$ s, the z-displacements of the tip of the projectile equal 21.15, 22.63, 24.63, 25.03 and 26.67 mm for P = 10, 25, 100, 200 and G.M. Zhang et al. / Composites: Part B 39 (2008) 476-489



Fig. 10. For pressure (from left to right) P = 10, 25, 100 and 200 MPa deformed shapes of the armor and the bullet at  $t = (a) 20 \mu s$ ,  $(b) 50 \mu s$ ,  $(c) 80 \mu s$ , and (d) 95 µs.

Ū.

300 MPa, respectively. Thus the axial displacement of the projectile increases with an increase in the clamping pressure applied to the frame bars.

Time histories of the axial velocity of the tail-end of the projectile are illustrated in Fig. 11. It can be seen that with an increase in the clamping pressure applied to the frame

K.



Fig. 11. For different clamping pressures applied to the frame bars time histories of the speed of the tail-end of the projectile.

bars, the speed of the residual projectile increases. Time histories of the kinetic energy and of the mass of the projectile are plotted in Fig. 12a and b. For a fixed value of time t the kinetic energy of the projectile increases noticeably with an increase in the clamping pressure. For P = 300 and 10 MPa, the kinetic energies of the residual projectile equal 206.84 and 80 J which, respectively, are 32.27% and 13.26% of the initial kinetic energy. The time history of the mass of the projectile, shown in Fig. 12b, suggests that more of the projectile has failed for smaller values of the applied pressure. Note that the speed of the residual bullet is also smaller for the lower value of the clamping pressure. Thus both the reduction in the mass and the reduction in the speed decrease the kinetic energy of the residual bullet when the clamping pressure is decreased. As mentioned above, for lower values of P, the armor held between the frames can move more easily towards the center of the frame and hence towards the bullet.

From the plot, shown in Fig. 13, of the time history of the x-displacement of the node A located at the center of the bottom left side of the armor enclosed in the frame bars (see Fig. 10), we conclude that for  $P \ge 200$  MPa, this node does not move due to the large frictional force between the armor and the frame. However, for smaller values of the pressure P, the frictional force is not large enough to prevent the armor from sliding between the frame bars and it moves toward the center more readily. For P = 10 MPa, the maximum x-displacement of this node is  $\sim 2.7$  mm.

Not only the armor enclosed in the frame bars moves towards the center, it is also compressed by the clamping pressure. Fig. 14 depicts the variation in the thickness of the armor between the frame bars versus time. For a fixed value of time, the change in the thickness of the armor is



Fig. 12. For different values of the clamping pressures applied to the frame bars, time histories of the (a) kinetic energy, and (b) the mass of the residual bullet.

not directly proportional to the applied pressure implying thereby that it undergoes both elastic and plastic deformations.

For the woven armor held in a four-bar frame with the clamping pressure of 200 MPa applied on it, Fig. 15 compares time histories of the tail-end speed and kinetic energy of the bullet when coefficients of friction between the frame bars and the armor, and that between any two adjacent layers are 0.3 and 0.2, respectively. With a decrease in the value of the coefficient of friction, the frictional force between the armor and the frame bars decreases causing the armor to move more easily towards the center. This consumes more of the kinetic energy of the bullet. Thus the effect of decreasing the coefficient of friction is similar to that of decreasing the clamping pressure applied to the frame bars.



Fig. 13. For different values of the clamping pressure time histories of the x-displacement of node A.



Fig. 14. For different values of the clamping pressure time histories of the distance between two opposite bars of the frame.

# 3.3. Armor held by uniform pressure applied to the two-bar frame

We now analyze deformations of the armor held in a two-bar frame; e.g. see Fig. 16. A uniform pressure is applied to the frame bars to hold the armor. Fig. 17 exhibits deformed configurations of the armor and the penetrator at t = 20, 50, 80 and 95 µs. These results are similar to those for the four-bar frame. The two free boundaries facilitate movement of the armor towards the frame center, and for t > 80 µs, yarns adjacent to one of the frame bars slip out of the frame bars; this slippage was not observed for the armor held in the four-bar frame.



Fig. 15. For the coefficients of friction equal to 0.2 and 0.3, time histories of the (a) speed of the tail-end, and (b) kinetic energy of the bullet. (The clamping pressure is 200 MPa.)



Fig. 16. A schematic sketch of the armor held in a two-bar frame.

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Fig. 17. For pressure P = 10, 25, 100 and 200 MPa (From left to right) deformed shapes of the armor and the bullet at (a)  $t = 20 \,\mu$ s, (b) 50  $\mu$ s, (c) 80  $\mu$ s, and (d) 95  $\mu$ s.

For different values of the applied pressure, time histories of the speed of the tail-end of the projectile are depicted in Fig. 18. As for the armor held in the four-bar frame, the speed of the residual bullet increases with an increase in the applied pressure, and it equals ~211,232,253,280 and 192 m/s, for P = 10,25,100,200 and 300 MPa, respectively.

We have plotted time histories of the kinetic energy and the mass of the projectile in Fig. 19a and b. The kinetic energy of the residual bullet increases significantly with an increase in the clamping pressure; nearly 18% more of the initial kinetic energy is consumed during the penetration process when the applied pressure is decreased from



Fig. 18. For a two-bar frame time histories of the speed of the tail-end of the projectile.

300 to 10 MPa. The time history of the y-coordinate of the bottom middle node B on armor's front surface (shown in Fig. 16) is given in Fig. 20. For P less than 200 MPa, the frictional force between the composite and the frame bars is not large enough to prevent sliding of the armor between the frame bars.

#### 3.4. Comparison of results for the two frames

We have compared in Table 5 the speed and the kinetic energy of the residual projectile for different pressures applied to a frame, and also for the same pressure imposed on the two-bar and the four-bar frames. It is clear that for a given value of the clamping pressure, the speed and the kinetic energy of the residual projectile are less when the armor is held in the two-bar frame than those for the four-bar frame. These trends become more vivid from the plots of Fig. 21a and b.

#### 3.5. Computation of the ballistic limit, $V_{50}$

We find  $V_{50}$  of the woven composite armor for the following four cases: 10 and 200 MPa clamping pressure applied to the two-bar and the four-bar frames. Table 6 gives the kinetic energy of the residual projectile for different initial speeds when the clamping pressure is 200 MPa and the armor is held in the four-bar frame. It is evident that for initial speeds greater than 275 m/s, the projectile penetrates the target completely. However, for initial speeds less than 270 m/s, the projectile is arrested in the target. Thus the ballistic limit, V<sub>50</sub>, of the woven armor held in the four-bar frame with a clamping pressure of 200 MPa is between 270 and 275 m/s. We have listed in Table 7 V<sub>50</sub> for the four cases. For both the two-bar and the four-bar frames, the V<sub>50</sub> decreases by about 30–45 m/s with a decrease in the clamping pressure from 200 to 10 MPa.



Fig. 19. For different values of the clamping pressure applied to the twobar frame, time histories of the (a) kinetic energy, and (b) mass of the projectile.

Fig. 22 evinces the reduction in the kinetic energy of the projectile versus its initial kinetic energy. For initial bullet speeds greater than the  $V_{50}$ , the kinetic energy absorbed increases almost linearly with an increase in the initial kinetic energy of the projectile, which agrees with Lim's [12] and Zeng's [13] result. However, for initial bullet speed less than 1.5 km/s our computations did not give a sudden additional reduction in the kinetic energy for initial bullet speeds greater than a critical value. It is possible that the critical speed of the bullet is greater than 1.5 km/s.

#### 3.6. Remarks

The element deletion technique used to simulate material failure may not realistically model material failure. Whereas numerical simulations indicate the Kevlar fiber



Fig. 20. For the two-bar frame, time histories of the y-displacement of node B for different values of the clamping pressure.

being broken into pieces that can act as projectiles, experimental observations suggest this not to be the case. Rather a Kevlar fiber is cut into two pieces in front of the projectile nose resulting in the formation of a pathway for the bullet. It seems that the node splitting technique such as that employed in [15,16] or the use of cohesive zones may be more appropriate for modeling the breakage of Kevlar fiber into two parts.

#### 4. Conclusions

We have numerically simulated three-dimensional deformations occurring during the penetration of a 9 mm FMJ, 124 grain projectile into soft body woven armor with the commercial finite element software, LS-DYNA. The projectile core is made of lead and is covered with a thin layer of copper. The geometry of the woven fabric is approximated by discretizing it into weft and warp yarns each of which is taken to be an orthotropic material. Frictional forces between adjoining layers and that between the armor and the frame bars are considered. Failed elements are deleted from the analysis.

We have delineated the effect of the frame size on the speed and the mass of the residual bullet, or equivalently on the fraction of the initial kinetic energy of the projectile dissipated during the penetration process. The effect of the frame size on the deformations of the projectile and the



Fig. 21. For the two-bar and the four-bar frames, the (a) speed and (b) kinetic energy of the residual projectile versus the clamping pressure.

armor has been ascertained by first regarding the frame bars to be rigid and the armor perfectly bonded to the square frame. Computed results reveal that for up to  $80.75 \text{ mm} \times 80.75 \text{ mm}$  frames the V<sub>50</sub> decreases with an increase in the frame size. However, we have not determined the minimum frame size so that for frame sizes greater than this value, the V<sub>50</sub> will be independent of the frame size.

Table 5

For different values of the clamping pressure, comparison of the speed and the kinetic energy of the residual bullet for the two-bar and the four-bar frames

Pressure (MPa)		10	25	100	200	300
Residual velocity (m/s)	Two-bar	192.36	210.69	231.98	253.47	280.28
	Four-bar	209.78	214.59	260.31	265.29	283.22
Residual kinetic energy (J)	Two-bar	68.96	86.44	115.68	149.11	186.91
	Four-bar	84.98	94.25	151.17	172.41	206.84

Table 6 The residual kinetic energy of the projectile for different initial velocity

Initial speed	Initial kinetic	Residual kinetic	Perforation
(111/5)	energy (J)	energy (J)	
600	1,442.17	759.73	Yes
400	640.97	167.02	Yes
350	490.74	96.43	Yes
300	360.55	15.95	Yes
280	314.07	10.03	Yes
275	302.96	1.25	Yes
270	292.04	0.00	No
260	270.81	0.00	No
250	250.38	0.00	No

Table 7

The hallistic limit for the four cases

Clamping pressure (MPa)		10	200
Ballistic limit, V <sub>50</sub> (m/s)	Two-bar frame	315-320	285-290
	Four-har frame	315-320	270-275



Fig. 22. The reduction in the kinetic energy of the projectile versus its initial kinetic energy.

We have also studied the effect on deformations of the armor and the projectile of the clamping pressure applied to the entire surfaces of the two- and the four-bar frames. It is found that an increase in the applied pressure reduces the kinetic energy of the bullet consumed during the deformation process. For the same applied pressure the two-bar frame is more effective in resisting the bullet than the fourbar frame.

The speed and the kinetic energy of the residual bullet decrease with a decrease in the coefficient of friction between the frame bars and the armor, and between adjacent layers of the armor. This effect is similar to decreasing the clamping pressure applied to the frame bars.

For the four-bar 42 mm  $\times$  42 mm frame clamped with a pressure of 200 MPa the V<sub>50</sub> is found to be  $\sim$ 270 m/s, and it decreases with an increase in the clamping pressure.

*Disclaimer:* Views expressed herein are those of authors and neither of the US Army nor of Virginia Polytechnic Institute and State University.

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# EFFECT OF MATRIX ON THE BALLSITIC PERFORMANCE OF SOFT BODY ARMORS

G. Gopinath<sup>1</sup>, J. Q. Zheng<sup>2</sup> and R. C. Batra<sup>1</sup>

<sup>1</sup>Department of Engineering Science & Mechanics, M/C 0219, Virginia Polytechnic Institute and State University, Blacksburg, VA 24061

<sup>2</sup>Program Executive Office – Soldier, US Army, 15395 John Marshall Highway, Haymarket, VA 20169, United States

#### Abstract

We study three dimensional deformations of a body armor composed primarily of Kevlar yarns and a small volume fraction of matrix, and account for their failure. The goal is to provide an improved understanding of different failure mechanisms and the effect of the volume fraction of the matrix material, and the bonding between the matrix and the yarn on the impact resistance of the armor.

#### 1. Introduction

Body armors made of woven fabric composites are extensively being used by the military and other law enforcement agencies to protect their personnel. Apart from preventing the projectile from penetrating, the vest must also be designed so that there is no noticeable bulge at the back face as this would lead to significant injuries even if the projectile does not completely penetrate the armor. The bulge height can be reduced by incorporating a layer of very soft fibrous material [1] between different armor layers. During penetration yarns which engage the projectile directly are called the principal or primary yarns. These yarns absorb most of the energy during impact and hence are the first to fail. Fibers possessing high strength and failure strain can absorb more energy before failing and hence form ideal candidates for use in fabric. Secondary yarns are those which do not come in direct contact with the projectile, and the amount of energy absorbed by these yarns is limited. It is intuitive to see that the ballistic performance of a body armor is improved if not only more yarns can engage the projectile during penetration but also disperse the stress waves away from the point of contact. Roylance [2] through numerical simulations showed that increasing friction between yarns leads to an increase in dispersion of stress waves. This was also shown to be true from experimental investigations by Briscoe and Motamedi [3] and through finite element simulations by Duan et al.[9]. Lee et al. [4] have studied the effect of matrix resin on the performance of fabric composites. Though the amount of matrix present in such composites is very small (typically in the range of 20-25% by volume) they can significantly influence the performance of the body armor. The presence of matrix has two important consequences; it not only restrains the yarns from moving but also helps in coupling different yarns together. Evidence for the above phenomena was given through a series of load deflection experiments and postmortem inspections conducted by Lee et al. [4]. Load deflection curves indicated that during penetration of the composite laminates there was a sudden drop in load after the failure whereas for armors made of only yarn fabric the load gradually dropped. The gradual load decrease was attributed to yarn slippage and successive breakage of individual yarns. Photographic evidence of the damaged area showed that more yarns were engaged for composites when compared to laminates made of only yarns. Also,

smaller penetration radius was observed for body armors made of only yarns as compared to laminates made of composites. Another consequence of having the matrix is that the effect of taper/curvature of the projectile on penetration is greatly reduced. It must be noted that the amount of energy absorbed by the resin material during penetration is only marginal. The above discussion may lead one to believe that the presence of matrix helps improve the ballistic performance, but this is not always the case as the matrix tends to make the body armor less flexible and hence the depth of the cone formed during penetration would be reduced leading to a less amount of energy absorbed. Also, the loss in flexibility can lead to reduced interaction between different layers of the fabric composite and it has generally been observed that laminates that have weak or no interaction tend to absorb less energy as compared to those that interact with each other [5,6,7]. Cheeseman and Bogetti [8] have suggested that very weak interaction between the matrix and yarn is preferable as it facilitates delamination between the matrix and the yarn allowing the fibers to extend to failure.

The presence of matrix on the ballistic performance of soft body armor has not been thoroughly studied in the literature, and conclusions have been drawn from results of a few experimental investigations such as that of Lee et al. [4]. The presence of matrix has two competing influences; on one hand it engages more yarns and prevents their sliding thereby increasing the ballistic performance of the body armor, on the other, it reduces the flexibility and the interaction among various layers thereby reducing the ballistic performance. We study this problem more comprehensively through numerical simulations as it is easier to assess results based on parameters that can be controlled. The problem is investigated by considering the impact of a hemispherical nosed cylindrical projectile consisting of a copper shell filled with lead on a woven composite made of Kevlar fabric and matrix resin. The effect of stiffness of the laminate on the ballistic performance is studied by considering polymers having different elastic moduli. The effect of bond strength between the matrix resin and the yarn fabric is also considered. The present investigation clearly shows how the matrix properties and the laminate stiffness significantly influence the overall performance of the body armor.

Though woven fabric has been previously modeled as one-dimensional (1D) or as 2D shell elements the use of 3D solid elements offers a more realistic view during the projectile penetration accounting for friction, and failure mechanisms etc. [10, 11]. Hence a more computationally intensive 3D finite element simulation using the commercial computer code LSDYNA has been carried out.

The rest of the paper is organized as follows. Section 2 describes the material and the geometric parameters of the armor and the projectile, constitutive relations, the failure criteria, and values assigned to different parameters. Results from simulations delineating effects of matrix properties on deformations and failure of the body armor are presented in Section 3.

#### 2. Material and geometric parameters

Commercial packages ABAQUS, ETA-VPG and LS-PREPOST were used for preprocessing the data. Figure 1 shows the woven composite with matrix resin and a representative volume element (RVE) of the composite laminate. The RVE exhibits distribution of the polymer matrix between the adjoining layers. Kevlar yarn bundle is modeled as a 3D continuum and meshed with eight-node brick elements. The width and the thickness of the yarn bundle were taken to be 1.0 mm and 0.75 mm, and no gap is assumed between yarn crossovers (this simplifies the geometric structure of the resin matrix and its finite element mesh). The volume fraction of the polymer in the RVE was found to be 15.5%. The polymer matrix was meshed with tetrahedral elements, and six layers of 75 mm x 75 mm composite laminates were used in the simulations. A gap of 0.1 mm is maintained between different layers and the overall thickness of the composite is about 9.5 mm. Simulations were performed by constraining the degrees of freedom of the end nodes of different layers of the laminates while the projectile penetrated the composite.

Material properties of the yarn are assumed to be strain rate independent since there is a lack of experimental data to realistically model this behavior. The material model MAT\_COMPSOTE\_DAMAGE available in LSDYNA is used to simulate the mechanical response of the yarn. Values of material properties taken from [10] are listed in Table 1.



Figure 1: Woven fabric composite, and a representative volume element

#### Table 1: Material parameters for the Kevlar yarn fabric

Mass density, $\rho = 1440 \text{ kg/m}^3$ ;
Elastic moduli, $Ea = 164.0 \text{ GPa}$ ; $Eb = Ec = 3.28 \text{ GPa}$ ;
Poisson's ratio, $\mu_a = \mu_b = \mu_c = 0;$
Shear moduli, $Gab = Gbc = Gca = 3.28$ GPa
Shear strength in the ab-plane $= 1.886$ GPa;
Longitudinal tensile strength along a-axis = 2.886 GPa;
Transverse tensile strength along b-axis = 1.486 GPa;
Transverse compressive strength along b-axis =1.7 GPa;
Normal tensile strength along $c$ -axis = 1.486 GPa;
Transverse shear strength in ca-plane = 1.586 GPa;
Transverse shear strength in cb-plane = 1.886 GPa.

Here a-axis is aligned along the yarn, b-axis transverse to the yarn in the plane of the layer, and the c-axis is along the normal to the ab-plane.

The yarn is assumed to fail at a principal strain of 0.2, and the option MAT\_ADD\_EROSION in LSDYNA is used to remove failed yarn elements from the calculations. This eliminates severely distorted elements from the finite element mesh, maintains reasonable time step size and enables simulations run for a longer time.

The polymer matrix is assumed to be isotropic and modeled using the material model MAT\_PIECEWISE\_LINEAR PLASTICITY in LSDYNA. We have specified the yield stress and the tangent modulus of the material. To see how the matrix properties affect the ballistic performance we have considered three sets of data to mimic a soft, moderately stiff and a very stiff polymer. Values of material parameters, listed below, have been arbitrarily chosen and may not correspond to any real material.

Mass density,  $\rho = 900 \text{ kg/m}^3$ ; Elastic modulus,  $E_1 = 0.5 \text{ GPa}$ ; Poisson's ratio,  $\mu_1=0.35$ ; yield stress,  $\sigma_{y1}=20\text{Mpa}$ Elastic modulus,  $E_2 = 3.5 \text{ GPa}$ ; Poisson's ratio,  $\mu_2=0.35$ ; yield stress,  $\sigma_{y2}=50\text{Mpa}$ Elastic modulus,  $E_3 = 7.0 \text{ GPa}$ ; Poisson's ratio,  $\mu_3=0.35$ ; yield stress,  $\sigma_{y3}=75\text{Mpa}$ 

The failure strain (plastic) for each matrix is taken to be 0.05, and the tangent modulus equal to one-half the elastic modulus. The Cowper-Symonds relation is used to include the effect of strain rate with

C= 4000s; P=0.182

To study the effect of matrix adhesion with the yarn we have considered two extreme cases. Tie constraints are imposed between the yarns and the matrix to represent perfect bonding and no constraints to represent no adhesion.

The projectile used is 13.3 mm long 9 mm Remington full metal jacket (FMJ), and is comprised of 0.5 mm thick outer copper layer with a solid lead shot filling. The values of

geometric and material parameters, identical to those used in Zhang et al. [10], are briefly summarized next. The Johnson–Cook (JC) relation incorporating damage is used to simulate the thermo-viscoplastic response of copper, and lead is modeled as an elastic perfectly plastic material; each material is assumed to be isotropic. Values assigned to different parameters are listed in Table 2.

Table 2: Values of material parameters for copper and lead in the penetrator

Values of material parameters for copper in
the JC constitutive relation
A = 0.09  GPa; B = 0.292  GPa; C = 0.025
n = 0.31; m = 1.09;
Mass density = $8950 \text{ kg/m}^3$
Specific heat = 385 J/kg K
Shear modulus = $47.27$ GPa
Bulk modulus = $102.4 \text{ GPa } D_1 1.0$
$D_2 = 1.0; D_3 = 0; D_4 = 0; D_5 = 0$
$Tm = 1356 \text{ K}; \sigma_{spall} = 1.9 \text{ GPa}$

#### Material parameters for lead

Mass density =  $11,340 \text{ kg/m}^3$ Elastic modulus = 16 GPaPoisson's ratio = 0.44Yield stress = 0.383 GPaFailure strain = 0.3

## 3.1 **Projectile impacting a single layer**

We first study the impact resistance of a single layer composite/yarn system to see how the matrix can influence its performance. Figure 2 exhibits the time history of the kinetic energy (KE) of the projectile for an initial velocity Vo = 100 m/sec; the soft and the stiff matrix refer to polymers with material properties indicated above by subscripts 1 and 3 respectively. It is evident that the three curves are close to each other signifying that the matrix mechanical properties have a little influence upon the decrease with time of the projectile's KE. The maximum decrease in the KE of the projectile for the time durations considered is for the composite with the stiff matrix and differs from that of the no matrix case by 12%. However, eventually at 200  $\mu$ s, the fiber only layer absorbs the most KE of the projectile. Also, there is only a marginal difference in the decrease in projectile's KE for composites containing the soft and the stiff matrices (for the cases being considered tieconstraints have been imposed between the matrix and the yarn). With time, however, the laminate made of only yarns continues to absorb energy before it fails whereas for the composite laminate failure starts to occur much earlier with limited amount of energy absorption. Thus the residual velocity of the projectile is much lower for the laminate made of only yarns as compared to the laminate having matrix. The primary reason for is that the addition of polymer makes the composite laminate less-flexible and it does not stretch as much as the laminate made of only yarns. This is also confirmed by comparing plots of the



Figure 2: Time history of the KE of the projectile impacting a single layer composite.

deformed laminates exhibited in Figure 3. The cone formed in the laminate made of only yarns is deeper than that in the laminate containing matrix resin. At 200  $\mu$ s the lamina made of only yarn has not completely failed so further reduction in the KE of the projectile is possible as compared to the laminate containing matrix in which the projectile has penetrated through the lamina. Even though the flexibility of the laminate made of only yarn has allowed it to absorb more energy, the cone formed is deeper too and that may adversely affect soldier's protection from getting injured.



a. Laminate made of only yarn b. Laminate made of yarn and matrix

Figure 3. Comparison of contour plots of the transverse displacement at 200 µs.

## 3.2 Projectile impacting a multi-layer composite system

Figure 4 shows the time history of the KE of the projectile as it penetrates a six-layer composite system. Cases marked adhesion and no-adhesion refer, respectively, to simulations in which tie-constraints have and have not been imposed between the matrix and the yarn. These results suggest that the residual velocity of the projectile decreases with an increase in the stiffness of the matrix, an enhancement in the bonding between the yarn and the matrix, and an increase in the material moduli of the matrix.



Figure 4. Time history of the KE of the projectile impacting a six-layer composite.







b. Total energy absorbed by the matrix

Figure 5. Time history of the total energy absorbed by the laminates

In order to understand how the matrix influences the performance of the body armor we have plotted in Figures 5a and 5b time histories of the total energy (TE = KE + IE) for all layers of yarn. The TE of laminates made of only yarns shows the greatest amount of energy absorbed during penetration. Also, for composites in which tie constraints have been imposed between the matrix and the yarn show less energy absorption as compared to that for composites in which no constraints have been imposed. This indicates that apart from restricting the motion of yarns during penetration the matrix in the composite also prevents the complete extension of the yarn before failure. This is also borne out by the fact that the matrix in those laminates which are strongly bonded to the yarns show marginally higher amount of energy absorption. As expected (see Fig. 6), laminates made of soft matrix material have a larger damage zone as compared to that in the laminate with stiff matrix material. In the absence of any matrix the number of yarns

which engage the projectile during penetration is limited to a small zone around the area of contact. The interaction between various yarns in this case is primarily through friction. The matrix resin allows for coupling between the yarns and hence a larger zone engages with the projectile. Apart from the coupling action, it also constrains the yarns from slipping. It is seen that the peak transverse displacement for laminates with no matrix is the largest, and it decreases with an increase in the stiffness of the matrix. Also, yarns being displaced are concentrated close to the point of contact for composites made of only yarn whereas for laminates with matrix the zone of influence is much larger and increases with the stiffness and the application of tie constraints.



Composites made of only yarn

a





It was found that composites having soft matrix performed poorly primarily because the matrix fails rather easily and hence is unable to couple different yarns together. Additionally, there is also a loss in flexibility due to the presence of matrix. This was verified by running a simulation in which the failure strain for the matrix was increased from 0.05 to 0.10. As evidenced by results exhibited in Fig. 8, there is a noticeable decrease in the KE of the projectile when the failure strain for the matrix is doubled.



Figure 8: Effect of increase in failure strain

## 4. Conclusions

The matrix in a composite significantly influences the ballistic performance of body armors because it decreases the flexibility of the armor and hence yarns are unable to stretch as much as they can without the matrix. A weak adhesion between the matrix and the yarn enables more yarns to stretch to their limiting values. However, the coupling and the constraining effects of the matrix on the yarns outweigh the loss in flexibility and improve body armor's performance. An increase in the material moduli of the polymer matrix enhances armor's performance. If trends seen in simulations with six layers hold when the number of layers is increased, then one would expect that the matrix would further magnify its influence on the ballistic performance for a composite with several layers.

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