

AD-A250 763



2

CONTRACTOR REPORT BRL-CR-689

# BRL

## PHYSICAL BASIS FOR A PRESSURE-DEPENDENT YIELD CRITERION FOR POLYMERS

RICHARD E. LYON  
LAWRENCE LIVERMORE NATIONAL LABORATORY

DTIC  
ELECTE  
MAY 28 1992  
S B D

MAY 1992

92-13995



APPROVED FOR PUBLIC RELEASE; DISTRIBUTION IS UNLIMITED.

U.S. ARMY LABORATORY COMMAND

BALLISTIC RESEARCH LABORATORY  
ABERDEEN PROVING GROUND, MARYLAND

## NOTICES

Destroy this report when it is no longer needed. DO NOT return it to the originator.

Additional copies of this report may be obtained from the National Technical Information Service, U.S. Department of Commerce, 5285 Port Royal Road, Springfield, VA 22161.

The findings of this report are not to be construed as an official Department of the Army position, unless so designated by other authorized documents.

The use of trade names or manufacturers' names in this report does not constitute indorsement of any commercial product.

# REPORT DOCUMENTATION PAGE

Form Approved  
OMB No. 0704-0188

Public reporting burden for this collection of information is estimated to average 1 hour per response, including the time for reviewing instructions, searching existing data sources, gathering and maintaining the data needed, and completing and reviewing the collection of information. Send comments regarding this burden estimate or any other aspect of this collection of information, including suggestions for reducing this burden, to Washington Headquarters Services, Directorate for Information Operations and Reports, 1215 Jefferson Davis Highway, Suite 1204, Arlington, VA 22202-4302, and to the Office of Management and Budget, Paperwork Reduction Project (0704-0188), Washington, DC 20503

<b>1. AGENCY USE ONLY (Leave blank)</b>		<b>2. REPORT DATE</b> May 1992	<b>3. REPORT TYPE AND DATES COVERED</b> Final, September 1987-January 1989	
<b>4. TITLE AND SUBTITLE</b>  Physical Basis for a Pressure-Dependent Yield Criterion for Polymers			<b>5. FUNDING NUMBERS</b>  PR: 1L162618AH80	
<b>6. AUTHOR(S)</b>  Richard E. Lyon				
<b>7. PERFORMING ORGANIZATION NAME(S) AND ADDRESS(ES)</b>  Chemistry and Materials Department Lawrence Livermore National Laboratory Livermore, CA 94550			<b>8. PERFORMING ORGANIZATION REPORT NUMBER</b>	
<b>9. SPONSORING / MONITORING AGENCY NAME(S) AND ADDRESS(ES)</b>  U.S. Army Ballistic Research Laboratory ATTN: SLCBR-DD-T Aberdeen Proving Ground, MD 21005-5066			<b>10. SPONSORING / MONITORING AGENCY REPORT NUMBER</b>  BRL-CR-689	
<b>11. SUPPLEMENTARY NOTES</b> The Contracting Officer's Representative for this report is Dr. Bruce P. Burns, U.S. Army Ballistic Research Laboratory, ATTN: SLCBR-IB-M, Aberdeen Proving Ground, MD 21005-5066.				
<b>12a. DISTRIBUTION / AVAILABILITY STATEMENT</b> Approved for public release; distribution is unlimited.			<b>12b. DISTRIBUTION CODE</b>	
<b>13. ABSTRACT (Maximum 200 words)</b>  A previously empirical pressure-dependent yield criterion for polymeric solids is derived from linear thermoelasticity and equilibrium thermodynamics. The assumption of a critical value of the mechanical component of the internal energy density at the onset of plastic deformation leads to an explicit expression for the yield condition in terms of the stress invariants and thermomechanical properties of the material. Order of magnitude agreement between the linear theory and published experimental data is obtained for a variety of polymers, suggesting a possible physical basis for pressure-dependent yielding.				
<b>14. SUBJECT TERMS</b>  Polymers, Stress Analysis, Pressure, Hydrostatic Pressure, Internal Energy			<b>15. NUMBER OF PAGES</b> 23	
			<b>16. PRICE CODE</b>	
<b>17. SECURITY CLASSIFICATION OF REPORT</b> UNCLASSIFIED	<b>18. SECURITY CLASSIFICATION OF THIS PAGE</b> UNCLASSIFIED	<b>19. SECURITY CLASSIFICATION OF ABSTRACT</b> UNCLASSIFIED	<b>20. LIMITATION OF ABSTRACT</b> UL	

INTENTIONALLY LEFT BLANK.

## TABLE OF CONTENTS

		<u>Page</u>
	ACKNOWLEDGMENTS .....	v
1.	INTRODUCTION .....	1
2.	THERMODYNAMIC YIELD CRITERION .....	3
3.	DISCUSSION .....	7
4.	REFERENCES .....	9
	DISTRIBUTION LIST .....	11



<b>Accession For</b>	
NTIS GRA&I	<input checked="" type="checkbox"/>
DTIC TAB	<input type="checkbox"/>
Unannounced	<input type="checkbox"/>
Justification	
By _____	
Distribution/	
<b>Availability Codes</b>	
Dist	Avail and/or Special
A-1	

INTENTIONALLY LEFT BLANK.

## ACKNOWLEDGMENTS

The author is indebted to Dr. R. M. Christensen, Dr. D. J. Nikkel, and Prof. R. J. Farris for helpful discussions and to the U.S. Army Ballistic Research Laboratory (BRL), Aberdeen Proving Ground, MD, for financial support. This work was performed under the auspices of the U.S. Department of Energy by the Lawrence Livermore National Laboratory, Livermore, CA, under Contract W-7405-Eng-48.

INTENTIONALLY LEFT BLANK.



## 1. INTRODUCTION

Structural applications for fiber-reinforced polymer composites demand that the thermoplastic or thermoset matrix materials survive complex multiaxial loading and severe environmental effects without causing failure of the composite part. Consequently, it is important to determine a failure or yield criterion for the constituent polymer matrix materials under a generalized stress state. This information in conjunction with modeling efforts on the state of stress in fiber-reinforced composites should provide insight into the poorly understood relationship between matrix- and composite-failure. Along these lines, the present work is an attempt to obtain a more basic understanding of multiaxial yielding in homogeneous polymeric solids.

In every relevant study to date, polymeric solids have been shown to exhibit significant increases in yield stress with pressure. For this reason, a yield criterion for polymers must include the effect of pressure or mean normal stress. One criterion which has the correct form to describe the observed variation of yield stress with pressure for polymeric solids is

$$J_2' = \sum_{i=0}^N \alpha_i J_1^i \quad (1)$$

which was originally proposed by Hu and Pae (1963). In Equation 1,  $J_2'$  is the second invariant of stress deviators,  $J_2' = 1/2 \sigma_{ij}' \sigma_{ij}'$  ( $i, j = 1, 2, 3$ ), and  $J_1$  the first stress invariant given by  $J_1 = \sigma_{kk} = \sigma_{11} + \sigma_{22} + \sigma_{33}$ . The deviatoric stresses are  $\sigma_{ij}' = \sigma_{ij} + p\delta_{ij}$ , for a mean normal pressure  $p = -1/3 \sigma_{kk}$ .

For  $N = 0$ , Equation 1 results in the distortional energy criterion proposed by Huber (1904) and Hencky (1924)

$$J_2' = \alpha_0 = k^2, \quad (2)$$

where the constant,  $k$ , is the yield stress determined in simple shear. Equation 2 has a physical basis in the assumption that a critical value of the distortional part of the strain energy determines the onset of plastic deformation (i.e., uniform pressure has no effect on yielding). Eichinger (1926) showed that the distortional energy criterion (Equation 2) is equivalent to the statement that yielding occurs when the shear stress on the octahedral plane,  $\tau_{oct}$ , reaches the critical value,  $\tau_{oct} \equiv (2J_2'/3)^{1/2} = 0.816 k$ . Von Mises (1913) assumed that a critical value of the mean difference in the principle stresses caused yielding and proposed that  $(J_2')^{1/2} = k$ . Although derived from different assumptions, each of the distortional energy, octahedral

shear stress, and von Mises criteria predict the same yield behavior and find great utility in describing the onset of plastic deformation in ductile metals at moderate pressures.

In contrast to metals, polymeric solids generally exhibit highly pressure-dependent yield behavior as evidenced by a significant increase in yield stress with hydrostatic pressure and a lower yield stress in simple tension than in simple compression. One way to account for these pressure effects is to expand  $J_2'$  as a polynomial in  $J_1$ , i.e.,

$$J_2' = k^2 + \alpha_1 J_1 \quad (3)$$

or

$$J_2' = k^2 + \alpha_1 J_1 + \alpha_2 J_1^2, \quad (4)$$

which correspond to Equation 1 with  $N = 1$  and  $N = 2$ , respectively. By analogy with Equations 2, Equations 3 and 4 are modified distortional energy yield criteria and have been used successfully to describe pressure-dependent yielding for a number of polymers (Mears, Pae, and Sauer 1969; Sauer, Mears, and Pae 1970).

Equation 3, in slightly different form (Raghava, Caddell, and Gregory 1973), was compared by Caddell, Raghava, and Adkins (1974) and Matsushige, Radcliffe, and Baer (1975) to a modified octahedral shear stress yield criterion for polymers proposed by Sternstein and Ongchin (1969), Bauwens (1970), and with minor variation by Spitzig and Richmond (1979). It was found that to first order in pressure, the inherently nonlinear, modified distortional energy criterion provided a generally better fit to the yield stress data for all of the polymers examined than did the modified octahedral shear stress/von Mises criterion—which predicts a linear variation of yield stress with pressure according to  $(J_2')^{1/2} = k + aJ_1$ . To obtain better agreement with high pressure data, Silano, Bhateja, and Pae (1979) expanded the von Mises criterion as a power series in  $J_1$  (i.e.,  $[J_2']^{1/2} = k + aJ_1 + bJ_1^2 + \dots$ ), achieving comparable predictive capability to the modified distortional energy criterion (Silano, Bhateja, and Pae 1974; Silano, Pae, and Sauer 1977; Pae 1977).

In contrast to the modified von Mises criterion, the modified distortional energy criterion maintains the dimensionality of a critical energy theory. Haigh (1970) proposed a total strain energy criterion for failure which includes both distortional and dilatational mechanical energy. In the above terminology, the total strain energy criterion takes the form:  $J_2' = k^2 + \alpha_2 J_1^2$ , which unlike Equation 3 or 4 does not contain

the first-order pressure term,  $\alpha_1 J_1$ , necessary to describe the observed variation of polymer yield stress with pressure. Consequently, within the context of the infinitesimal, equilibrium theory, a mechanical energy criterion cannot account for polymer yielding under a generalized state of stress. Recognizing that mechanical energy is only one form of free energy, the following sections examine the possibility that a different thermodynamic potential—the internal energy—may be the criterion for polymer yielding.

## 2. THERMODYNAMIC YIELD CRITERION

For a reversible solid deformation process, the combined first and second laws of thermodynamics can be written for a closed system at uniform temperature,  $T$ ,

$$dU = TdS + \sigma_{ij} d\epsilon_{ij}, \quad (5)$$

where  $dU$ ,  $dS$  are incremental changes in the internal energy and entropy per unit volume, respectively, and  $\epsilon_{ij}$  are the components of the small strain tensor

$$\epsilon_{ij} = \frac{1}{2} \left[ \frac{\partial u_i}{\partial x_j} + \frac{\partial u_j}{\partial x_i} \right]. \quad (6)$$

Following Sneddon (1974), we consider that  $U = U(\epsilon_{ij}, T)$ , so that the total differential for internal energy is written

$$dU = \left( \frac{\partial U}{\partial \epsilon_{ij}} \right)_T d\epsilon_{ij} + \left( \frac{\partial U}{\partial T} \right)_{\epsilon_{ij}} dT. \quad (7)$$

The Helmholtz free energy,  $A$ , is defined

$$A = U - TS. \quad (8)$$

Writing Equation 8 in incremental form and substituting Equation 5,

$$dA = \sigma_{ij} d\epsilon_{ij} - SdT, \quad (9)$$

from which the following relationships are obtained:

$$\left(\frac{\partial A}{\partial \epsilon_{ij}}\right)_T = \sigma_{ij}; \quad \left(\frac{\partial A}{\partial T}\right)_{\epsilon_{ij}} = -S, \quad (10)$$

showing that

$$\left(\frac{\partial \sigma_{ij}}{\partial T}\right)_{\epsilon_{pq}} = -\left(\frac{\partial S}{\partial \epsilon_{ij}}\right)_T. \quad (11)$$

From Equations 5 and 11,

$$\left(\frac{\partial U}{\partial \epsilon_{ij}}\right)_T = \sigma_{ij} - T \left(\frac{\partial \sigma_{ij}}{\partial T}\right)_{\epsilon_{pq}}. \quad (12)$$

In addition, it can be shown that

$$\left(\frac{\partial U}{\partial T}\right)_{\epsilon_{ij}} = T \left(\frac{\partial S}{\partial T}\right)_{\epsilon_{ij}} = C_\epsilon, \quad (13)$$

where  $C_\epsilon$  is the heat capacity of the solid at constant strain. Substituting Equations 12 and 13 into Equation 7 gives the variation in internal energy density for incremental changes in strain and temperature as

$$dU = \left[ \sigma_{ij} - T \left(\frac{\partial \sigma_{ij}}{\partial T}\right)_{\epsilon_{pq}} \right] d\epsilon_{ij} + C_\epsilon dT, \quad (14)$$

which provides the necessary relationship to calculate the internal energy change for a solid deformation if the constitutive equation for the material is known.

The constitutive relationship for an isotropic, linear thermoelastic solid can be written

$$\sigma_{ij} = \frac{E}{(1+\nu)(1-2\nu)} [\nu \epsilon \delta_{ij} + (1-2\nu)\epsilon_{ij} - (1+\nu)\beta(T - T_o)\delta_{ij}], \quad (15)$$

with  $E$  and  $\beta$  the Young's modulus and linear thermal expansion coefficient of the material having Poisson's ratio,  $\nu$ . The dilatation,  $e$ , is  $e = \delta V/V_o = \epsilon_{ii}$ . Substitution of Equation 15 into Equation 14 and integrating gives the internal energy for a linear thermoelastic solid, in general, as

$$U - U_o = \frac{1}{2} \sigma_{ij} \epsilon_{ij} + \frac{E\beta(T + T_o)e}{2(1 - 2\nu)} + C_\epsilon(T - T_o) \quad (16)$$

where  $U_o$  denotes the internal energy per unit volume of the reference state,  $\epsilon_{ij} = 0$ ,  $T = T_o$ . The second term on the right-hand side of Equation 16 results from the coupling of elastic deformation and thermal diffusion, while the third term represents the heat content. In contrast to the free energy, the internal energy includes thermoelastic coupling even at the reference temperature,  $T = T_o$ .

The first term on the right-hand side of Equation 16 is the strain energy density which can be decomposed into distortional and dilatational components via

$$\frac{1}{2} \sigma_{ij} \epsilon_{ij} = G \epsilon'_{ij} \epsilon'_{ij} + \frac{1}{2} K \epsilon_{ii} \epsilon_{kk} \quad (17)$$

with  $G$  and  $K$  the shear and bulk modulus, respectively. The deviatoric strain is  $\epsilon'_{ij}$ , which is defined  $\epsilon'_{ij} = \epsilon_{ij} - e/3 = \epsilon_{ij} - \epsilon_{ii}\delta_{ij}/3$ . Since  $\sigma'_{ij} = 2G\epsilon'_{ij}$ , and  $\sigma_{kk} = 3K\epsilon_{ii}$ , Equation 17 can be written as

$$\frac{1}{2} \sigma_{ij} \epsilon_{ij} = \frac{1}{4G} \sigma'_{ij} \sigma'_{ij} + \frac{\sigma_{kk}^2}{18K} \quad (18)$$

Thermoelastic coupling is reflected in the second term on the right-hand side of Equation 16 which can be expressed as

$$\frac{E\beta(T + T_o)e}{2(1 - 2\nu)} = \frac{1}{2} \beta(T + T_o) \sigma_{kk} \quad (19)$$

Substituting Equations 18 and 19 into Equation 16 results in an expression for the internal energy change for an isothermal deformation in terms of the stress components

$$U - U_o = \frac{1}{4G} \sigma'_{ij} \sigma'_{ij} + \frac{1}{2} \beta(T + T_o) \sigma_{kk} + \frac{\sigma_{kk}^2}{18K} + C_\epsilon(T - T_o) \quad (20)$$

In terms of the stress invariants, the internal energy change takes the form

$$U - U_o = \frac{1}{2G} J_2' + \frac{1}{2} \beta (T + T_o) J_1 + \frac{1}{18K} J_1^2 + C_e (T - T_o) . \quad (21)$$

The assumption is now made that the criterion for yielding of an isotropic polymeric solid in a state of combined stress is that the mechanical part of the internal energy density of the material reaches some critical value. In other words, we neglect the heat content and of the material and consider only that part of the internal energy which involves mechanical coupling with the outside world. Partitioning the internal energy according to  $U - U_o = \Delta + C_e (T - T_o)$ , the mechanical part of the internal energy,  $\Delta$ , is obtained from Equation 21 as

$$\Delta = \frac{1}{2G} J_2' + \frac{1}{2} \beta (T + T_o) J_1 + \frac{1}{18K} J_1^2 . \quad (22)$$

Considering only the mechanical internal energy as contributing to failure eliminates the unlikely possibility that a pure heating process can cause yielding in the absence of stress.

The critical value of the mechanical internal energy at yield can be determined from a simple shear experiment where  $J_1 = 0$ , so that  $J_2' = \sigma_{12}^2 = \tau_y^2$  at the onset of plastic deformation, and Equation 22 becomes

$$\Delta = \tau_y^2 / 2G = k^2 / 2G . \quad (23)$$

The condition for isothermal yielding at temperature,  $T$ , in a general stress state is finally obtained from Equations 22 and 23 as

$$J_2' = k^2 - [G\beta(T + T_o)] J_1 - [G/9K] J_1^2 , \quad (24)$$

which is the empirical Equation 4, with the pressure coefficients now defined as  $\alpha_1 = -G\beta(T + T_o)$  and  $\alpha_2 = -G/9K$ .

The implication of a critical value of the mechanical internal energy density at yield can be demonstrated graphically for a uniaxial deformation. If the coordinate axes are chosen such that the

stresses are in the principle directions, a uniaxial tension or compression in the 1 direction has  $\sigma_{11} = \pm\sigma_1$ ,  $\sigma_{22} = \sigma_{33} = 0$ , so that Equation 22 can be written

$$\Delta = \frac{1}{2} \beta (T + T_o) \sigma_1 + \frac{1}{2E} \sigma_1^2 \quad (25)$$

The quadratic relationship between mechanical internal energy and uniaxial stress for an isothermal deformation is shown schematically in Figure 1. Stipulating a critical value for the mechanical internal energy at yield requires that the magnitude of the yield stress in simple compression be larger than the yield stress in simple tension. This mean normal pressure effect on uniaxial yielding is described by Equation 24.

The predicted negative sign of the  $J_1$  coefficient,  $\alpha_1$  in Equation 24 is in agreement with every relevant experimental study of polymer yielding. The magnitude of  $\alpha_1$  for engineering thermoplastics at  $T = T_o = 295^\circ K$  can be calculated using literature values for  $\beta$  and  $G$  (van Krevelen 1976) to be of the order  $\alpha_1 = -2G\beta T_o \approx -25 \pm 10$  MPa, which is roughly 10 times larger than typical experimental values of  $\alpha_1 \approx -3 \pm 2$  MPa (Mears, Pae, and Sauer 1969; Sauer, Mears, and Pae 1970; Raghava, Caddell, and Gregory 1973; Caddell, Raghava, and Adkins 1974; Matsushige, Radcliffe, and Baer 1975; Sternstein and Ongchin 1969; Bauwens 1970; Spitzig and Richmond 1979; Silano, Bhateja, and Pae 1974; Silano, Pae, and Sauer 1977; Pae 1977) obtained for various thermoplastics. The magnitude of the  $J_1^2$  coefficient,  $\alpha_2$ , is estimated to be small and on the order of  $\alpha_2 = -G/9K \approx -.02$ , which compares favorably with the measured  $\alpha_2 = -.010$  and  $-.017$  for polyethylene and polypropylene, respectively (Mears, Pae, and Sauer 1969). However, values (Sauer, Mears, and Pae 1970) for polytetrafluoroethylene ( $\alpha_2 = 1.2 \times 10^{-5}$ ) and polycarbonate ( $\alpha_2 = 5.1 \times 10^{-4}$ ) are positive and significantly smaller than predicted by Equation 24.

### 3. DISCUSSION

The physical basis for a pressure-dependent yield criterion for polymers may be a critical value of the mechanical part of the internal energy density at the onset of plastic deformation. With this assumption, an isothermal yield condition is obtained from linear thermoelasticity and equilibrium thermodynamics in terms of the stress invariants, temperature, and material properties which has the same form as an experimentally proven polynomial expansion to second order in pressure.

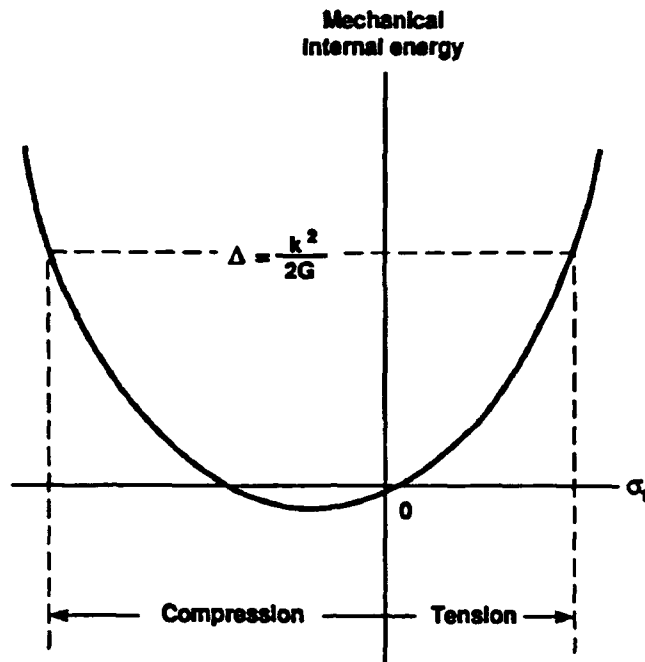


Figure 1. Schematic Showing Relationship Between Mechanical Internal Energy and Uniaxial Stress for an Isothermal Deformation.

Order of magnitude agreement between predicted and published experimental values of the pressure coefficients is observed. This result is encouraging, considering the highly nonlinear, nonequilibrium behavior of typical engineering thermoplastics prior to yielding and the characteristically large strains ( $\epsilon \approx 4$  to 5%) at which fully plastic deformation commences. Moreover, the linear theory does not address the universally observed pressure dependence of the shear, tensile (Sauer 1977; Pae and Bhateja 1975), and bulk (Bhateja and Pae 1975) moduli, nor the possibility of premature failure due to inhomogeneous, localized deformation (Stemstein, Ongchin, and Silverman 1968).

It can be shown (Lyon and Farris 1987) that, for an isothermal deformation at the reference temperature, the linear term in  $J_1$  appearing in Equation 24 corresponds to the reversible heat of deformation (i.e., the entropic component of the internal energy). Considering the statistical nature of macromolecules, it is reasonable that a successful yield criterion for polymer solids should reflect the entropic origin of chain deformation.



#### 4. REFERENCES

- Bauwens, J. C. Journal of Polymer Science. A-2, vol. 8, p. 893, 1970.
- Bhateja, S. K., and K. D. Pae. Journal of Macromol. Sci.-Revs. Macromol. Chem. Vol. C13, p. 77, 1975.
- Caddell, R. M., R. S. Raghava, and A. G. Adkins. Material Science Engineering. Vol. 13, p. 113, 1974.
- Eichenger, A. Proceedings 2nd. Ind. Congress Applied Mechanics. Zurich, p. 325, 1926.
- Haigh, B. P. Engineering. Vol. 109, p. 158, 1920.
- Hencky, H. z. fur Angew. Math. Mechanik. Vol. 4, no. 4, p. 323, 1924.
- Hu, L. W., and K. D. Pae. Journal of the Franklin Institute. Vol. 275, p. 491, 1963.
- Huber, M. T. Czasopismo techniczne, Lemberg (Polish). Vol. 22, p. 81, 1904.
- Lyon, R. E., and R. J. Farris. Polymer. Vol. 28, p. 1127, 1987.
- Matsushige, K., S. V. Radcliffe, and E. Baer. Journal of Material Science. Vol. 10, p. 883, 1975.
- Mears, D. R., K. D. Pae, and J. A. Sauer. Journal of Applied Physics. Vol. 40, p. 4229, 1969.
- Pae, K. D. Journal of Material Science. Vol. 12, p. 1209, 1977.
- Pae, K. D., and S. K. Bhateja. Journal of Macromol. Sci.-Revs. Macromol. Chem. Vol. C13, p. 1, 1975.
- Raghava, R., R. M. Caddell, and S. Y. Gregory. Journal of Material Science. Vol. 8, p. 225, 1973.
- Sauer, J. A. Polymer Engineering and Science. Vol. 17, p. 150, 1977.
- Sauer, J. A., D. R. Mears, and K. D. Pae. European Polymer Journal. Vol. 6, p. 1015, 1970.
- Silano, A., S. K. Bhateja, and K. D. Pae. International Journal of Polymer Material. Vol. 3, p. 117, 1974.
- Silano, A., K. D. Pae, and J. A. Sauer. Journal of Applied Physics. Vol. 48, p. 4076, 1977.
- Sneddon, I. N. The Linear Theory of Thermoelasticity. Ch. 1, Springer-Verlag, NY, 1974.
- Spitzig, W. A., and O. Richmond. Polymer Engineering and Science. Vol. 19, p. 1129, 1979.
- Sternstein, S., and L. Ongchin. American Chemical Society Div. Polymer Chem. Polym. Preper. Vol. 10, p. 1177, 1969.
- Sternstein, S. S., L. Ongchin, and A. Silverman. Applied Polymer Symposia. No. 7, p. 175, Interscience, NY, 1968.

van Krevelen, D. W. Properties of Polymers. Elsevier, Amsterdam, 1976.

von Mises, R. Akademie der Wissenschaften, Gottingen, Mathematisch-Physikalische Klasse. Nachrichten, 1913.

<u>No. of Copies</u>	<u>Organization</u>	<u>No. of Copies</u>	<u>Organization</u>
2	Administrator Defense Technical Info Center ATTN: DTIC-DDA Cameron Station Alexandria, VA 22304-6145	1	Commander U.S. Army Tank-Automotive Command ATTN: ASQNC-TAC-DIT (Technical Information Center) Warren, MI 48397-5000
1	Commander U.S. Army Materiel Command ATTN: AMCAM 5001 Eisenhower Ave. Alexandria, VA 22333-0001	1	Director U.S. Army TRADOC Analysis Command ATTN: ATRC-WSR White Sands Missile Range, NM 88002-5502
1	Commander U.S. Army Laboratory Command ATTN: AMSLC-DL 2800 Powder Mill Rd. Adelphi, MD 20783-1145	1	Commandant U.S. Army Field Artillery School ATTN: ATSF-CSI Ft. Sill, OK 73503-5000
2	Commander U.S. Army Armament Research, Development, and Engineering Center ATTN: SMCAR-IMI-I Picatinny Arsenal, NJ 07806-5000	2	Commandant U.S. Army Infantry School ATTN: ATZB-SC, System Safety Fort Benning, GA 31903-5000
2	Commander U.S. Army Armament Research, Development, and Engineering Center ATTN: SMCAR-TDC Picatinny Arsenal, NJ 07806-5000	(Class. only) 1	Commandant U.S. Army Infantry School ATTN: ATSH-CD (Security Mgr.) Fort Benning, GA 31905-5660
1	Director Benet Weapons Laboratory U.S. Army Armament Research, Development, and Engineering Center ATTN: SMCAR-CCB-TL Watervliet, NY 12189-4050	(Unclass. only) 1	Commandant U.S. Army Infantry School ATTN: ATSH-CD-CSO-OR Fort Benning, GA 31905-5660
(Unclass. only) 1	Commander U.S. Army Rock Island Arsenal ATTN: SMCRI-TL/Technical Library Rock Island, IL 61299-5000	1	WL/MNOI Eglin AFB, FL 32542-5000  <u>Aberdeen Proving Ground</u>
1	Director U.S. Army Aviation Research and Technology Activity ATTN: SAVRT-R (Library) M/S 219-3 Ames Research Center Moffett Field, CA 94035-1000	2	Dir, USAMSAA ATTN: AMXSY-D AMXSY-MP, H. Cohen
1	Commander U.S. Army Missile Command ATTN: AMSMI-RD-CS-R (DOC) Redstone Arsenal, AL 35898-5010	1	Cdr, USATECOM ATTN: AMSTE-TC
		3	Cdr, CRDEC, AMCCOM ATTN: SMCCR-RSP-A SMCCR-MU SMCCR-MSI
		1	Dir, VLAMO ATTN: AMSLC-VL-D
		10	Dir, USABRL ATTN: SLCBR-DD-T

<u>No. of Copies</u>	<u>Organization</u>	<u>No. of Copies</u>	<u>Organization</u>
11	<p>Director Benet Weapons Laboratory U.S. Army Armament Research, Development, and Engineering Center ATTN: SMCAR-CCB, J. Keane T. Allen J. Vasilakis G. Friar J. Zweig L. Johnson T. Simkins V. Montvori J. Wrzochalski G. D'Andrea R. Hasenbein Watervliet, NY 12189</p>	1	<p>Commander U.S. Army Belvoir RD&amp;E Center ATTN: STRBE-JBC, C. Kominos Fort Belvoir, VA 22060-5606</p>
		2	<p>Commander U.S. Army Laboratory Command Harry Diamond Laboratories ATTN: SLCHD-TS-NT, A. Frydman 2800 Powder Mill Road Adelphi, MD 20783-1197</p>
		1	<p>Commander U.S. Army Laboratory Command ATTN: AMSLC-TD, R. Vitali Adelphi, MD 20783-1145</p>
9	<p>Commander U.S. Army Armament Research, Development, and Engineering Center ATTN: SMCAR-CCH-T, S. Musalli P. Christian K. Fehsal N. Krasnow R. Carr SMCAR-CCH-V, E. Fennell SMCAR-CCH, J. DeLorenzo SMCAR-CC, R. Price J. Hedderich Picatinny Arsenal, NJ 07806-5000</p>	1	<p>Commander U.S. Army Missile Command ATTN: AMSMI-RD, W. McCorkle Redstone Arsenal, AL 35898</p>
		3	<p>Commander U.S. Army Armament Research, Development, and Engineering Center ATTN: SMCAR-FSA-M, R. Botticelli F. Diorio SMCAR-FSA, C. Spinelli Picatinny Arsenal, NJ 07806-5000</p>
		3	<p>Project Manager Advanced Field Artillery System ATTN: COL Napoliello LTC D. Ellis G. DelCoco Picatinny Arsenal, NJ 07806-5000</p>
2	<p>Commander U.S. Army Armament Research, Development, and Engineering Center ATTN: SMCAR-TD, V. Linder T. Davidson Picatinny Arsenal, NJ 07806-5000</p>	1	<p>Commander Watervliet Arsenal ATTN: SMCWV-QA-QS, K. Insko Watervliet, NY 12189-4050</p>
1	<p>Commander Production Base Modernization Activity U.S. Army Armament Research, Development, and Engineering Center ATTN: AMSMC-PBM-K Picatinny Arsenal, NJ 07806-5000</p>	2	<p>Project Manager SADARM Picatinny Arsenal, NJ 07806-5000</p>

<u>No. of</u> <u>Copies</u>	<u>Organization</u>	<u>No. of</u> <u>Copies</u>	<u>Organization</u>
8	Project Manager Tank Main Armament Systems ATTN: SFAE-AR-TMA, COL Hartline SFAE-AR-TMA-MD, C. Kimker H. Yuen R. Joinson J. McGreen SFAE-AR-TMA-ME, K. Russell D. Guzowitz SFAE-AR-TMA-MP, W. Lang Picatinny Arsenal, NJ 07806-5000	1	Commander David Taylor Research Center Ship Structures and Protection Department ATTN: J. Corrado, Code 1702 Bethesda, MD 20084
2	PEO-Armaments ATTN: SFAE-AR-PM, D. Adams T. McWilliams Picatinny Arsenal, NJ 07806-5000	4	Director Lawrence Livermore National Laboratory ATTN: R. Christensen S. deTeresa W. Feng F. Magness P.O. Box 808 Livermore, CA 9455
2	Commander Wright-Patterson Air Force Base ATTN: AFWAML, J. Whitney R. Kim Dayton, OH 45433	2	Pacific Northwest Laboratory A Div of Battelle Memorial Inst. Technical Information Section ATTN: M. Smith M. Garnich P.O. Box 999 Richland, WA 99352
1	Commander DARPA ATTN: J. Kelly 1400 Wilson Blvd. Arlington, VA 22209	6	Director Sandia National Laboratories Applied Mechanics Department, Division-8241 ATTN: C. Robinson G. Benedetti W. Kawahara K. Perano D. Dawson P. Nielan P.O. Box 969 Livermore, CA 94550-0096
2	Director U.S. Army Materials Technology Laboratory ATTN: SLCMT-MEC, B. Halpin T. Chow Watertown, MA 02172-0001	1	Director Los Alamos National Laboratory ATTN: D. Rabern MEE-13, Mail Stop J-576 P.O. Box 1633 Los Alamos, NM 87545
1	Commander Naval Research Laboratory ATTN: I. Wolock, Code 6383 Washington, DC 20375-5000	3	University of Delaware Center for Composite Materials ATTN: J. Gillespe B. Pipes M. Santare 201 Spencer Laboratory Newark, DE 19716
2	Commander David Taylor Research Center ATTN: R. Rockwell W. Phyllaier Bethesda, MD 20054-5000		

<u>No. of Copies</u>	<u>Organization</u>
2	North Carolina State University Civil Engineering Department ATTN: W. Rasdorf L. Spainhour P.O. Box 7908 Raleigh, NC 27696-7908
1	University of Utah Department of Mechanical and Industrial Engineering ATTN: S. Swanson Salt Lake City, UT 84112
1	Stanford University Department of Aeronautics and Aeroballistics Durant Bldg. ATTN: S. Tsai Stanford, CA 94305
1	Pennsylvania State University Department of Engineering Science and Mechanics ATTN: T. Hahn 227 Hammond Bldg. University Park, PA 16802
1	University of Illinois at Urbana-Champaign National Center for Composite Materials Research 216 Talbot Laboratory ATTN: J. Economy 104 South Wright Street Urbana, IL 61801
1	University of Illinois at Urbana-Champaign Department of Aeronautical and Astonautical Engineering ATTN: A. Zak Urbana, IL 61801
2	Olin Corporation Flinchbaugh Division ATTN: E. Steiner B. Stewart P.O. Box 127 Red Lion, PA 17356

<u>No. of Copies</u>	<u>Organization</u>
1	Olin Corporation ATTN: L. Whitmore 10101 9th St., North St. Petersburg, FL 33702
3	Alliant Techsystems, Inc. ATTN: C. Candland J. Bode K. Ward 5640 Smetana Dr. Minnetonka, MN 55343
1	Chamberlain Manufacturing Corporation Research and Development Division ATTN: T. Lynch 550 Esther Street P.O. Box 2335 Waterloo, IA 50704
1	Custom Analytical Engineering Systems, Inc. ATTN: A. Alexander Star Route, Box 4A Flintstone, MD 21530
2	Institute for Advanced Technology ATTN: T. Kiehne H. Fair 4030-2 W. Braker Lane Austin, TX 78759

USER EVALUATION SHEET/CHANGE OF ADDRESS

This laboratory undertakes a continuing effort to improve the quality of the reports it publishes. Your comments/answers below will aid us in our efforts.

1. Does this report satisfy a need? (Comment on purpose, related project, or other area of interest for which the report will be used.) \_\_\_\_\_

2. How, specifically, is the report being used? (Information source, design data, procedure, source of ideas, etc.) \_\_\_\_\_

3. Has the information in this report led to any quantitative savings as far as man-hours or dollars saved, operating costs avoided, or efficiencies achieved, etc? If so, please elaborate. \_\_\_\_\_

4. General Comments. What do you think should be changed to improve future reports? (Indicate changes to organization, technical content, format, etc.) \_\_\_\_\_

BRL Report Number BRL-CR-689 Division Symbol \_\_\_\_\_

Check here if desire to be removed from distribution list. \_\_\_\_\_

Check here for address change. \_\_\_\_\_

Current address: Organization \_\_\_\_\_  
Address \_\_\_\_\_

**DEPARTMENT OF THE ARMY**  
Director  
U.S. Army Ballistic Research Laboratory  
ATTN: SLCBR-DD-T  
Aberdeen Proving Ground, MD 21005-5066



NO POSTAGE  
NECESSARY  
IF MAILED  
IN THE  
UNITED STATES

OFFICIAL BUSINESS

**BUSINESS REPLY MAIL**  
FIRST CLASS PERMIT No 0001, APG, MD

Postage will be paid by addressee.

Director  
U.S. Army Ballistic Research Laboratory  
ATTN: SLCBR-DD-T  
Aberdeen Proving Ground, MD 21005-5066

