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INTERACTIVE EFFECTS OF HIGH- AND LOW-FREQUENCY LOADING ON FATIGUE

MECHANICAL TECHNOLOGY INCORPORATED 968 ALBANY-SHAKER ROAD LATHAM, NEW YORK 12110

May 1985

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MATERIALS LABORATORY AF WRIGHT AERONAUTICAL LABORATORIES AIR FORCE SYSTEMS COMMAND WRIGHT-PATTERSON AFB, OHIO 45433

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FOREWORD

This report was prepared by Mechanical Technology Incorporated (MTI), Latham, New York for the Metals Behavior Branch, Metals and Ceramics Division, Materials Laboratory, Air Force Wright Aeronautical Laboratories, (AFWAL/MLLN), Wright-Patterson Air Force Base, under Contract F33615-82-C-5056. The work was administered under the direction of Dr. David I.G. Jones, AFWAL/MLLN. The program effort was conducted at MTI by A. Petrovich, W.F. Bessler and W.H. Ziegler. Additional support was provided by J. Walton, L.A. Peterson and L. Isley. This report describes work conducted from September, 1982 to December, 1984.



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I. INTRODUCTION

A means of more fully utilizing the useful life of aircraft engine components is provided by the Retirement-for-Cause (RFC) life management concept. Under the RFC philosophy, components are inspected at intervals of operation such that a crack or other service induced defect just below the level of detectability cannot grow to a critical size between inspections. Those components with no observable flaw are returned to service with the assurance that if a fatigue crack develops, it will not grow to a size that will result in catastrophic failure of the component while in service. A cost savings results from the fact that by retiring the components on the condition of an observable flaw, components without flaws that would otherwise be retired by the probabalistic scheme would now be allowed to remain in service until cracking is apparent.

Implementation of the Retirement-for-Cause method requires advances both in non-destructive evaluation and crack growth life prediction. The primary requirement with regard to crack growth life prediction is an improvement in the accuracy of life prediction for the complex loading profile experienced by engine components. Important to the improvement of the accuracy of life prediction is an accounting of the interaction between the various components of the loading profile.

The life limiting loading profile experienced by an engine disk consists basically of a group of low frequency cycles associated with thermal gradients or centrifugal forces and superimposed high frequency loading associated with blade passage. The cycle period associated with the low frequency cycle (low cycle) loading is on the order of seconds to several hundred seconds. A wide range of loading rates and load levels may also be involved in the low cycle loading. The high frequency cycle (high cycle) loading would typically involve frequencies on the order of hundreds to several thousand hertz. Important to accurate life prediction is establishing the manner in which each of these features of the engine disk loading profile contribute to crack growth and how these features interact. The specific aspects of combined cycle loading that must be addressed are the following:

- Establishment of the limits of high cycle loading under which the disk can be safely operated.
- How cumulative damage rules should be applied when combined high cycle/low cycle loading contribute to crack growth.
- The degree to which the high cycle and low cycle loading influence each others contribution to crack growth.

A test system was designed and constructed specifically for the present study to provide adequate load levels up to 2000 Hz and minimize the frequency ranges over which dynamic complications in load application are present. A purely servo-hydraulic system based on an Akashi voice-coil servo-valve was used for all of the testing. The load frame and specimen were designed to minimize the number of system resonances that create undesirable specimen stress patterns and either complicate or invalidate the representation of stresses around the specimen crack. The test system constructed for their study is described in Section 2.

The specimen type used for this study was a center crack panel also described in Section 2. A clevis arrangement with provisions to clamp the specimen ends was used to grip the specimen. By securely clamping the specimen and providing additional lateral support, specimen resonances could be avoided at the selected test frequencies. Both the high frequency and low frequency was sensed by a load cell. It was recognized that resonances in the load frame and specimen could disturb the correlation between the load cell measurement and stresses in the specimen as well as provide significant bending stresses associated with resonant lateral vibration. Modal analyses of a preliminary specimen were performed to determine its natural frequencies and mode shapes over a range of steady load and crack length. These modal analyses indicated the specimen modifications required to make the specimen suitable for testing in bands of frequencies up to 2000 Hz. Test frequencies of 200 and 1825 Hz were chosen for most of this study. The absence of excessive bending stresses and a proper correlation between load cell measurement and specimen stresses was verified at these frequencies with strain gage measurement on the specimen. The precision in the high frequency ΔK measurement required for this study made these specimen dynamic evaluations and detailed verification of specimen stress absolutely essential.

The high/low frequency loading profile used in this study is described in Section 2. The low frequency component was a trapezoidal waveform with a rise time (T_1) and fall time (T_2) of 0.5 seconds and with a hold time (T_0) of between 2 and 180 seconds. The high frequency loading was applied during the low frequency cycle hold period and typically ranged between 220 and 4450 Newtons (50 to 1000 lbs). The low frequency load levels P₁ and P₂ were varied during the tests such that the low frequency stress intensity factors K₁ and K₂ were maintained constant. The high frequency load range (P_0) was either increased during the test or maintained constant which in either case resulted in an increasing high frequency stress intensity factor range (K_0) . The low cycle R ratio (P_1/P_2) was 0.1 for all of the testing. All testing was performed at 649°C.

Section 3 presents the results of a series of crack growth tests that were performed on Inconel 718 at $649^{\circ}C$ ($1200^{\circ}F$). The following aspects of the high/low cycle interaction were investigated in the series of crack growth tests:

- the effect of high cycle frequency up to 2000 Hz on the low cycle/high cycle interaction
- the effect of low cycle stress intensity factor range (ΔK_{LC})
- variation of crack growth rate as a function of high cycle stress intensity factor range (ΔK_{HC}) over a ΔK_{LC} range of 15 to 40 MPa \checkmark m
- the influence of low cycle hold time between 2 and 180 seconds on the crack growth rate under combined cycle loading

The results of testing are summarized in curves representing crack growth rate versus high frequency ΔK for constant low frequency cycle ΔK range and low cycle hold time. Crack growth rate is reported in terms of growth per unit time at the upper level of the low cycle trapezoidal loading profile. The low cycle ΔK ranges included in the testing were 15, 20, 30 and 40 MPa \checkmark m. The low cycle hold times included 2, 5, 10 and 180 seconds. Comparisons are made in Section 3 between the results for a high cycle frequency of 200 and 1825 Hz provided by this study and those for 10 Hz provided in Reference 1.

Section 4 also explores possible mechanisms associated with the combined cycle interaction. Correlations are made between the crack growth data and features

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of the fracture surface. Means ofdelling combined cycle crack growth rate are also discussed.

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IL SYSTEM CONSTRUCTION, EVALUATION OF SYSTEM DYNAMICS, AND SPECIMEN DESIGN

A. Background on the Selection of Test Equipment

There have been several approaches to providing controllable load levels in the frequency regime above 100 Hz. One system used for high frequency fatigue testing is the electrodynamic shaker. Motion and forces in these systems are generated by interaction of a solenoid generated field with a moveable armature. Standard commercial electrodynamic shakers have force ratings up to 9000 lbs. with up to 100 g's of acceleration available to 3000 Hz.⁽²⁾ Magnetrostrictive devices have also been used to generate forces and displacements of frequencies up to 1000 Hz. An example of a study of threshold crack growth conducted with magnetostrictive system is that of Reference 3.

The desire to test materials with a loading profile similar to that of an aircraft turbine engine has lead to the development of test systems that can apply low cycle high amplitude loading (on the order of 5000-20,000 lbs.) along with low amplitude high frequency (100 to 2000 lbs.) loading. A novel example of the machines developed for the application of combined cycle loading is the major/minor cycling system constructed by Instron Ltd.⁽⁴⁾ The characteristics of this system are summarized in Table 1 and a schematic representation of the system appears in Figure 1. The high frequency loading component is applied by the electrodynamic shaker and the low frequency component by a hydraulic actuator. This is made possible by a specially developed isolation unit between the hydraulic actuator and shaker which allows the simultaneous application of loading by the hydraulic actuator and electrodynamic shaker.

An alternative to a combined servohydraulic/electrodynamic and purely electrodynamic system for the application of a combined high cycle/low cycle loading profile is a purely servohydraulic system based on a voice coil servo-valve.⁽⁵⁾ The use of electrohydraulic servo-valves for material testing is widely practiced and is usually performed with a flapper-nozzle valve, which, in spite of its inherent low-frequency limitation, has been totally adequate for the testing of such material properties as creep, ultimate strength, yield strength, and low-frequency cyclic fatigue.

TABLE 1

SPECIFICATION FOR INSTRON COMBINED CYCLE TEST SYSTEM

Trapezoidal

0.1 - 99.9 secs. or 0.1 - 99.9 min.

0.4 sec.

50kN

700mm

661 x 305

4 ±250kN

High Frequency Component:

Waveform: Frequency Range: Max. Dynamic Load: Sinusoidal 50 - 600Hz depending upon the specimen stiffness ±5kN

Low Frequency Component;

Waveform: Minimum rise and fall times: Dwell times: Maximum Load Unidirectional tension or compression:

Load Frame:

Number of Columns: Dynamic Load Rating: Max, Vertical Daylight: (between load cell & shaker) Distance between columns:

Load Cell:

Dynamic:

Fatigue Rating (Unidirectional) : Excitation: Load Measurement Accuracy Static: 50kN max. force 5.6 volt DC ±1% of indicated force or ±0.2% of full scale, whichever is the greater ±3% of indicated force or ±0.2% of full scale, whichever is the greater

Compensation is provided for changes in dynamic load reading caused by the mass of the Grip or Fixture.

NOTE:-

Because of the high operating frequencies, the mass of the moving parts has a significant effect on the performance of the machine. The actual frequency range over which the desired dynamic force can be achieved is dependent upon the stiffness of the specimen. Details of the specimen should be given when ordering.

Patents Pending

Instron Limited reserves the right to change details and specifications without notice

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Figure 1 Schematic of Major/Minor Cycling Machine Control System (4)

An example of a high frequency servo-hydraulic system is that developed by Akashi Ltd.⁽⁵⁾ The Akashi vibration system employs a servo-valve and actuator with high-frequency capability based on a voice-coil-type servo-valve. In this system, the electrical drive signal directly causes servo-valve spool motion. The voice-coil valve thereby provides a significant advantage in high frequency input flow capability to the actuator. Also the Akashi servo-valve is optimized to reduce the impedance loading associated with high frequency, and its mechanical natural frequency has been established to favor frequencies at the higher end of its useful spectrum. The two types of servo-valves, the flapper-nozzle system and Akashi voice-coil system, are shown in Figures 2 and 3 respectively. Additional high frequency servo-hydraulic systems are manufactured by MTS⁽⁶⁾ and Teem.

The ability of a test system to provide adequate displacement at the test frequency is the most important consideration for high frequency testing. It is difficult to accurately estimate the displacement capability of a shaker in specimen fatigue testing application in view of the complexity of the interaction between the actuator and load frame. A means of establishing a rough estimate is to determine the maximum deflection capability of the shaker with a test load of 50 lbs. and without the constraint of a load frame. Such a determination was made for an Akashi test system based on a pilot/slave servo-valve with a 5 gpm/37gpm flow capability and a 1.2 inch stroke actuator. The estimated deflection of this system is shown in Figure 4. This curve provided an adequate estimate of deflection for planning fatigue testing in a test system based on this servo-valve/actuator combination.

Another important consideration for fatigue crack growth testing is ensuring that the stresses around the growing crack are similar to those under quasistatic conditions. Only if this can be assured can results from one frequency be compared to another. Resonance in the load frame and specimen often disturb the patterns of stresses. Investigations of the manner in which specimen stresses can be distorted at high frequency have been carried out with the aid of modal analysis.⁽⁷⁾ Figure 5 shows the patterns of standing wave resonant vibration that can occur in a standard center crack panel specimen. Such dynamic complications to crack growth testing can be minimized by proper selection and design of the test system along with proper design of the specimen.

Figure 2 Standard Flapper-Nozzle Servo-Hydraulic Valve

84144

Boost System - Steady State with Spool Off Center





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Magnel



Figure 3 Hydraulic Actuator and Servo Valve (5)

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Figure 4 Estimated Maximum Displacement versus Frequency for the Akashi Servo-valve servo-actuator combination HV 10.0-1.2-37/5.



Figure 5 Vibration patterns for a center panel specimen (7)

B₂ System Selection and Construction

The study of the phenomenon of high-frequency/low frequency load interaction in fatigue and creep crack growth required the construction of a test system that could provide adequate levels of load up to a frequency of 2000 Hz that could provide low cycle load levels up to 10,000 lbs., and allow precise control of both low-frequency and high-frequency loading during fatigue crack growth test-ing. The following options were available.

- A purely electrodynamic system with a maximum force rating on the order of 10,000 lb.
- A combined high-frequency electrodynamic/low-frequency servo-hydraulic system with some type of isolation system to remove large preloads from the electrodynamic system during high-frequency vibration.
- A purely servo-hydraulic system based on a voice-coil servo-valve.

A preliminary evaluation indicated that the purely electrodynamic and purely servo-hydraulic options would permit a load frame sufficiently rigid to preclude dynamic complications at the higher frequencies. It was also determined that a servo-hydraulic system was less expensive for the load ranges required for this program. It was feared that the combined servo-hydrualic and electrodynamic system with its isolation system had such an extended load frame that a large number of resonances would make testing and verification of loading extremely difficult particularly at higher frequencies. Therefore, because of compatibility with existing MTI equipment, relatively low cost, and ability to construct compact frames and fixturing around the specimens and actuator, the purely servo-hydraulic option was chosen.

The purely servo-hydraulic test system constructed for this program is based on an Akashi voice-coil servo-valve with a frequency capability that far exceeds the more conventional flapper-nozzle servo-valve. The Akashi servo-actuator incorporates all the features of the fatigue-rated actuators currently used today, as well as several key design improvements. The piston-to-cylinder clearance is manufactured to allow operation without a piston seal, thereby eliminating a wear item and, more importantly in high-frequency operation, removing the cause of waveform distortion associated with seal motion during pressure reversal. Hydrostatic bearings are employed to provide high side load capability and to eliminate any metal-to-metal contact at the bearings. Piston rod seals are not used, thus eliminating the largest factor of friction in the system and also removing an element that must be periodically serviced, i.e., replacing the seal and refinishing the piston rod.

The high-frequency servo-hydraulic equipment purchased by MTI for this program includes the following:

- Akashi Servo-Actuator Model HV 10.0-1.2-37/5 (10,000 lb. maximum status load
- Akashi Servo-Valve (pilot/slave) Model SV 5/SV 37
- Akashi Servo-Controller Model SC-1
- Akashi Servo-Amplifier Model SA-400
- Akashi Manifold Model HM-40

The Akashi servo-actuator (Model HV 10.0-1.2-37/5) was installed in a load frame already in operation at MTI. Figure 6 shows the servo-hydraulic testing system with specimen, induction heating coil, and crack-length-measuring telemicroscope in place. Figure 7 shows, in greater detail, the specimen, induction heating coil and specimen gripping arrangement. A standard clevis was used to provide loading to the specimen. To reduce unnecessary deflection, the 2-in.-wide center cracked panels are reinforced at the ends gripped by the clevis.

During the high-frequency experiments, noise levels reached 135 db, and it was necessary to construct a sound-deadening enclosure around the system. The noise reduction provided by this enclosure was sufficient to reduce the noise to an acceptable level in surrounding offices and work areas.

The high-frequency servo-valve receives signals from the servo-controller and servo-amplifier. The command signal applied to the servo-controller is the sum of a high- and low-frequency signals which are controlled independently by the PDP-11/04 computer that interfaces the servo-hydraulic system. A schematic diagram of the control and data acquisition system is shown in Figure 8. The high- and low-frequency signals must be controlled independently becasue the system gain (i.e., the load range per unit input signal) is different for the high- and low-frequency portions of the command signal. Figure 9 schematically



Figure 6 High Frequency Servohydraulic Test System with 2" Width Center Cracked Panel Specimen Installed



Figure 7 Center-Cracked Panel Specimen in High-Frequency Test System (The specimen is heated inductively, and the crack length is monitored by both a 20X telemicroscope and an AC direct potential crack-measurement system.)







Figure 9 Closed-Loop Control System for High- and Low-Frequency Components of Loading Profile

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illustrates the control of the high- and low-frequency signals. The loading profile resulting from the summation of the high- and low-frequency components is shown in Figure 10a which corresponds to the load response of the system with the 2-in.-wide center-cracked panel and a high-frequency component of 460 Hz. The high-frequency segment of the load cell signal is shown in greater detail in Figure 10b.

A servo-hydraulic vibration system test report for the Akashi servo-valve and servo-actuator is included in Appendix A. The first chart in this report shows the maximum performance in terms of acceleration for the Akashi servo-actuator used in this test program. This chart provides the acceleration as a function of frequency for a 60.75 and 108.7 kg payload. The second and third charts in this report show the acceleration as a function of frequency for a specific signal level input to the servo-amplifier for a 60.75 kg and 108.7 kg payload respectively. Such charts are useful in estimating the actuator displacement capability in a materials testing load frame.

The basic types of specimens used in crack growth testing are the compact tensile specimen, the edge-cracked-panels, the center-cracked panel and bend-type specimens, plus many variations on these basic types. For high-frequency testing, several important dynamic considerations have to be addressed if the desired loading conditions are to be achieved at the crack. The most important consideration is the compliance (extension per unit load) for the range of ΔK (K = stress intensity factor) to be covered in the test program. The specimen has to provide as low a compliance as possible consistent with accurate measurement of crack length and of crack growth rate. (Minimizing compliance is important because, as frequency increases, the possible deflection of both servo-hydraulic and electrodynamic systems rapidly decrease). Therefore, the center-cracked-panel type was chosen for use in this program because of it relatively low compliance for a given ΔK range.

In the planning of the test program, frequency ranges for the high and low-frequency ΔK were selected. frequencies. A specimen and loading system were then selected to accomplish testing in these ranges. The dynamic limitations of the system had to be considered. For example, in the frequency regime of 1000 Hz and above, the maximum displacement that a servo-hydraulic system can provide is critically dependent on the design of the servo-valve and servo-actuator. Simi-



Figure 10a Combined high/low frequency loading profile.





lar considerations apply to purely electrodynamic and combined servo-hydraulic/electrodynamic systems. The specimen dimensions and starting crack length (a) were, therefore, chosen based on system displacement limitations and desired high-frequency ΔK range.

C. Dynamic Evaluation of Test Specimens

Strain gage measurements on a preliminary center crack specimen (Figure 11 and 12) used in the test program showed that while it was adequate for 200 Hz testing it was inadequate for frequencies above 500 Hz. A series of system resonances resulted in substantial bending stresses and a generally poor correlation between load cell measurements and strain gage measurements of stress in the crack region. A modal analysis was performed on this preliminary specimen to determine how it should be modified to reduce the density of resonances in the frequency regime beyond 500 Hz. The resulting modifications on the specimens were successful in providing several frequency bands above 500 Hz in which undesirable dynamic stresses were eliminated.

An important objective of the present work is to compare the fatigue crack growth behavior of the material at several frequencies. It is, therefore, extremely important to ensure that the loading pattern is the same at these frequencies. A series of specimen designs were evaluated with strain gages and a successful design evolved. The criteria established for the specimen were as follows:

- Bending stresses and out of phase components of stress at the specimen crack line must be less than 5% of the high cycle amplitude (this is a requirement of ASTM standard E647 extended to cover all dynamic d combines of stress uniformity).
- The isad measurement must have an appropriate relationship to the stresses in the crack region throughout the test.
- Stresses at several locations in the crack stress field must have an appropriate relationship to each other.

In view of the fact that resonant frequencies shift as the crack grows, the following condition must also be fulfilled by the specimen:





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Figure 12 Drawings Showing the Dimensions of the Preliminary Specimen Used for 200 Hz Testing

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- The above conditions must apply over a range of at least 200 Hz in the trequency regime of interest over the entire range of loading and crack length experienced during a test.

The interval of 200 Hz was chosen based on the expected variation in resonant frequency resulting from specimen material modulus variation as the specimen is brought to the test temperature. This can be demonstrated by considering the expression for the ringing frequency of an undamped plate:

$$v_{n} = \frac{1}{2\pi} \kappa_{n}^{2} \left(\frac{EIg}{\gamma S}\right)^{1/2}$$
(2.1)

where κ_n is determined by the boundary conditions,

and where

S = specimen cross-sectional area
Y = density
E = elastic modulus
I = moment of inertia
g = accelerometer of gravity
Q = length of specimen

With the average 10% reduction in elastic modulus over the specimen region participating in the vibration mode, a 5% change is expected in resonant frequency or about 100 Hz at a resonant frequency of 2000 Hz.

In Sections C.l and C.2, the results and conclusions of the modal analysis are presented along with a description of the specimen adopted for testing around 2000 Hz and a summary of the strain gage dynamic evaluation of this specimen at 1825 and 2000 Hz.

C.1 Modal Analysis of the Preliminary Specimen

After the evaluation of the preliminary specimen (shown in Figures 11 and 12) with strain gages, it was apparent that some modification of the specimen would be required to make it suitable for testing near 2000 Hz. Comparison of strain gage response from opposite surfaces indicated that bending stresses well beyond that permitted by ASTM standard E647 existed over most of the frequency range
from 1000 to 2000 Hz. It was felt that a modal analysis should be performed on the specimen to determine the resonant frequencies and mode shapes of these resonances. The information gained from this modal analysis was subsequently used to determine modifications that would be necessary to make the specimen suitable for 2000 Hz testing.

The preliminary specimen with the clamping fixtures extended (Figure 13) was placed in the load frame in the usual manner and a static load was applied. An accelerometer was attached to the specimen or load frame at one of 43 locations. The specimen was then excited with a random signal having the prectrum shown in Figure 14. The signal was provided with a Scientific Atlanta Vibration Controller which has the capability of open loop or closed loop vibration control. The modal analysis was conducted with open loop excitation. The accelerometer was moved successively to the locations on the specimen shown in Figure 15 and the random vibration was applied. Additional locations on the load frame were included but the levels of vibration were considerably less than those on the specimen. The accelerometer response and shaker excitation were simultaneously recorded. The data was processed using a Hewlett Packard 5451C Fourier Analyzer System which calculates the transfer function between the input and response at points on the specimen and load frame. An analytical model was curve fitted to the transfer function data and modal parameters such as natural frequency, damping factor, and mode shape were identified. The system software also has the capability of providing animated representations of the mode shapes. Modal parameters for the following mean load and crack length (.'a) cases were evaluated:

- Mean load = 2000 lbs., crack length (2a) = 0.20"
- Mean load = 4500 lbs., crack length (2a) = 0.20"
- Mean load = 2000 lbs., crack length (2a) = 0.95"

Table 2 lists the natural frequencies and damping factors for the resonant modes for each of these cases. The mode shapes for the three cases are shown in Figure 16 through 18.

The modul analysis on this specimen demonstrated the following:



Figure 13 Drawings Showing the Dimensions of the Preliminary Specimen Subjected to Modal Analysis







Figure 15 Accelerometer locations used in the modal analysis

MODE	NAT. FREQ. (HZ)	DAMP. FACT. (%)	DAMP. COEFF. (RAD/SEC)
1	765.0115	.8677	41.7039
2	1439.8879	3.0185	273.2087
3	1709.9106	1.4115	151.6672
4	1861.4131	1.5402	180.1682
5	1955.4885	.4472	54.9416

CASE I: Mean load = 2000 lbs, total crack length (2a) = 0.20"

CASE II: Mean load = 4500 lbs, total crack length = 0.20"

MODE	NAT. FREQ.	DAMP. FACT. (%)	DAMP. COEFF. (RAD/SEC)
1	827.7138	3.6762	191.3148
2	1483.2148	1.0132	94.4276
3	1673.6045	.9470	99.5824
4	1806.6233	2.1325	242.1211
5	1953.5444	.3507	43.0522

CASE III: Mean load = 2000 lbs, total crack length = 0.95 "

MODE	NAT. FREQ. (HZ)	DAMP. FACT. (%)	DAMP. COEFF. (RAD/SEC)
1	646.3500	.3708	15.0568
2	795.7371	2.7693	138.5117
3	1381.7729	3.9457	342.8291
4	1690.2246	1.4620	155.2824
5	1883.5137	1.4233	168.4624



Figure 16 Mode Shapes for Case I: Mean Load 2000 Lbs, Crack Total Length (2a) of 0.20"



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Figure 17 Mode Shapes for Case II: Mean Load 4500 Lbs, Crack Total Length (2a) of 0.20"



Figure 18 Mode Shapes for Case III: Mean Load 2000 Lbs, Crack Total Length of 0.95"

- The load level and crack length changes expected during a test can shift resonant frequencies substantially, in some cases as much as 200 Hz.
- While these resonances involve the entire load train and possibly the frame as well, the most severe lateral deflection is in the specimen.
- There was a cluster of resonances from 1400 to 2000 Hz with an average of 200 Hz spacing between them.
- There was a rather clear field between 750 Hz to 1200 Hz where no resonances developed for any of the cases.

The behavior shown in the modal analysis corresponds well to that shown by strain gage response of this preliminary specimen. Strain measurements confirmed that there is a regime between approximately 800 to 1200 Hz in which there are no complications from dynamic bending stresses. In the regime between 1200 to 2000 Hz, strain measurements likewise confirmed the series of resonances that result in significant bending strains that cannot be tolerated in a test. In several of the mode shapes the greatest deflection is in the thin part of the specimen. It was felt that if these deflections can be reduced significantly by lateral support and damping, the specimen may provide a satisfactory stress distribution at the crack region.

C.2 Strain Gage Evaluation of a Specimen with Lateral Reinforcement and Damping

The strain gage evaluation of the preliminary specimen showed that bending strains were excessive over most of the frequency between 500 and 2000 Hz with the possible exception of the range 1000 to 2000 Hz. The ranges of frequency over which the strains on either side of the specimen are in phase and of equal magnitude was extremely limited and certainly less than the 200 Hz that is required due to shifting resonances during a test.

Recognizing that the preliminary specimen with the clamping arrangement extended as much as possible was still unsuitable for testing above 1200 Hz, several experiments were performed to determine the effectiveness of various approaches to spreading out or damping resonant vibrations. The modal analyses showed that the most extreme deflection is in the unclamped portion of the specimen. It appears that since this region is the most compliant it acts somewhat as a hinge between the load train elements above and below it. Three additional modifications were, therefore, made to the specimen and system:

- the construction of lateral buttressing that would reinforce the specimen in the unclamped region with respect to out of plane motion yet have minimal effect on tensile stress distribution
- removal of elements in the load train which may add to unwanted deflections
- increasing the size of the clamping fixture to improve stiffness in the lateral direction

Figure 19 shows the lateral reinforcements applied to the specimen. A schematic diagram of the central region of the specimen indicating the elements of the lateral support are shown in Figure 20. Parallel elements were loaded against the unclamped surfaces of the specimen near the crack. By applying this load through several layers of glass cloth, the support still had substantial compliance in shear and would not significantly affect the distribution of tensile stresses in the specimen. The glass cloth was also expected to provide damping of lateral vibration. Specific dimensions of the lateral support and clamping fixtures are shown in Figure 21.

A series of experiments were conducted on this specimen to establish its suitability and also to establish a set of procedures for verifying an appropriate stress distribution on each specimen prior to each experiment. The set of experiments involved strain gage measurements in the locations shown in Figure 22.

The first group of experiments involved the amplitude measurement of two strain gages on opposite surfaces when a 2500 lb. preload and a high cycle amplitude were applied. Figure 23, for example, shows the output of strain gage 1 and 2 as a function of frequency. Over most of the frequency range the correlation is acceptable indicating a satisfactorily low level of bending stresses at the crack line over most of this frequency range. There are, however, several frequency ranges in which large discrepancies occur. The stress amplitudes as shown by strain gages 1 and 7 on the same side of the specimen (Figure 24) likewise show a good correlation for this uncracked specimen over most of the



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Figure 19 Specimen with End Reinforcement and with Lateral Constraints Applied to the Crack Region



Figure 20 Diagram Showing Location of Damping Blocks and Glass Insulating Material



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Figure 21 Dimensions of Specimen End Clamps



Figure 22 Diagram Showing Location of Strain Gages





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frequency range investigated but with significant discrepancies in certain narrow ranges of frequency.

The effect of reducing the length of the load train by removing the load cell and compression rings was evaluated. As shown in Figure 25, the effect of eliminating these objects was to modify the ranges over which differences in amplitude between strain gages 1 and 2 occurred. While the correlation between these outputs is improved in the frequency range between 1900 and 2200 Hz, in other ranges the correlation showed little change and in some cases deterioration. It was found in subsequent tests that removing the compression rings alone can improve the correlation of strain gage measurements in the frequency range between 1900 and 2200 Hz.

This series of experiments, involving the measurement of stresses at various locations on the specimen, demonstrated that there are frequency ranges above 1000 Hz in which dynamic stresses do not dist rb the stress pattern associated with tensile loading. Satisfied that the specimen with lateral damping could provide satisfactory test results in some frequency band near 2000 Hz experimentation was carried further to identify frequency bands in which testing could be carried out and also establish procedures that could be used to verify the appropriateness of the stress distribution prior to each test. The verification on each specimen is necessary because there is a possibility that the effectiveness of the lateral support may depend on the procedures of assembly.

The verification procedures adopted involved the measurement and comparison of both phase and amplitude on opposite surfaces of the specimen. An initial experiment was carried out over a range of frequencies near 2000 Hz to determine how the relative magnitude and phase vary with changing load and crack length. The strain measurements were along the crack line at locations 1 and 2 of Figure 22. Measurement of the output of strain gages 1 and 2 were made and displayed on an oscilloscope as 1 versus 2. In the absence of bending stresses the resultant would be a line at 45° from the x or y axis, i.e. the stresses would be in phase and of equal magnitude. Resonant vibrations are apparent as a deviation from this pattern. An appropriate stress distribution would have a maximum peak to peak deviation of 5% from the ideal 45° trace. This condition would also be required over a 200 Hz interval around the chosen test frequency in order to ensure that resonances are not "swept in" by increasing the specimen temper-





it acceptable level from speciment to specimen.

Prior to each crack growth, test strain gage measurements were made in this manner over a 200 Hz frequency interval to ensure that the specimen was properly assembled. This procedure was adopted to ensure that errors in assembly that might reduce the effectiveness of the lateral support had not occurred.

The coat sensing for the high frequency load range was performed with the remote load cell. For frequencies near 2000 Hz it was required to apply a correction tatter to the measured load in order to properly represent the stresses in the strength and applied head was measured as a function of strain gage output at 20 and 200 Hz. The propertionality at these two frequencies was consistently the same. The propertionality between load cell output and strain gage output was then measured at 1825 and 2000 Hz. The correction factor that must be applied to load cell measurement in order to provide the same proportionality as at the lower trequencies was established for the range of load and crack length measurement experienced in a typical experiment. The correction factor variation was 12% over a typical range of test conditions.

Sensing load directly on the specimen at locations 7 or 8 shown in Figure 22 was considered. However, in view of the fact that elevated temperature strain gages would be required and that strain gages on the specimen are frequently destroyed near 2000 Hz, it was decided to perform tests with remote load sensing. With remote sensing a much higher testing productivity was achieved with perhaps a small sacratice of absolute high frequency load measurement accuracy.











FIGURE 28 Series of Oscilloscope Representations of Strain Gage Output #1 versus #2 Showing the Degree of Bending and Out of Phase Vibration on the Specimen Crack Centerline for a Mean Load of 2000 lbs. and a Crack Length of 0.700".

III. EXPERIMENTAL TEST PROGRAM

The test program objective was the establishment of the relationship between crack growth rate in an aircraft engine disk material and those parameters associated with the high- and low-frequency loading experienced in the engine firstand second-stage disks. The information provided by this testing program was to be applicable to life prediction of flawed engine components and provide guidance to the implementation of the retirement-for-cause engine maintenance and design concept. Consequently, the test parameters selected for this program were based on the loading conditions experienced by aircraft engine disks.

Considering the nature of the loading and the requirements of aircraft engine design, the following were included in the experimental program:

- Determination of the nature of the transition from low cycle to high cycle dominated behavior over ranges of both low cycle and high cycle stress intensity factor range (ΔK).
- Establishment of the high-cycle transition ΔK over as wide a frequency range as possible.
- Major cycle (low cycle) hold times in the regimes in which both fatigue and creep crack growth dominate. Cycle times from a few seconds to several hundred seconds were included.
- A temperature typical of those experienced by the aircraft engine disk. For the Inconel 718 specimens used in this study, 1200°F was chosen.
- A sufficient level of replication to eliminate the influence of material variability on the test results and indicate the level of consistency of the experimental system.
- The influence (if any) of the high-cycle loading on crack growth below the high-cycle transition.

The test program summarized in Table 3 was designed to address these aspects of high- and low-frequency interaction in crack growth. The low cycle waveform used throughout the testing program was a trapezoidal loading profile with ramp times of 0.5 seconds and hold times ranging from a second to essentially infinity (steady mean load). The high-frequency loading was applied during the hold period only.

	TABLE 3	
TES	T PROGRAM OUTLINE	
(ALL	TESTING AT 1200°F)	

Objective	Conditions of Test
Evaluate high-cycle threshold ΔK and crack growth rate versus high frequency ΔK in the creep crack growth regime.	Selected low-frequency ΔK values with long hold times (60 seconds or greater) maintained throughout the test and with varying high-frequency ΔK values.
Evaluate high-cycle threshold Ak and crack growth rate versus high-frequency Ak in the fatigue crack growth regime.	Selected low-frequency ΔK values with the shortest prac- tical hold times (probably on the order of 1 to 10 seconds) maintained throughout the test and with varying high-frequency ΔK values.
Evaluate the effect of high- frequency loading on low- cycle crack growth in the re- gime of transition between creep and fatigue-dominated crack propagation.	Selected low-frequency hold times with specific low- frequency ΔK levels maintained throughout the test and with varying high-frequency ΔK levels.
Evaluate the effect of high- frequency loading at several low-frequency cycle R ratios in the creep and fatigue crack growth regimes.	Varying high-frequency ΔK values emphasizing the high- cycle transition regime with specific low cycle hold times maintained throughout the test.
Evaluate the effect of temp- erature on the high-cycle transition.	Varying high-frequency ΔK values emphasizing the transition high- cycle regime with specific low- frequency ΔK and hold times and selected temperatures.

A. Fatigue Crack Growth Studies Conducted at 200 Hz

Combined cycle tests with a high cycle trequency of 200 Hz were conducted for low cycle parameters in a test matrix. All testing in this matrix was carried out with a low cycle R ratio of 0.1 and a test temperature of $1200^{\circ}F$. This matrix included low cycle maximum K values ranging from 15 to 40 ksi \checkmark in (corresponds to a ΔK of 15 to 40 MPa \checkmark m since the R ratio was 0.1) and low cycle hold times ranging from 2 to 180 seconds. Table 4 shows the conditions of tests completed and the number identifying the test. All conditions in the test matrix were applied in at least one test. In several cases replicated tests were conducted. Data plots for all of these tests may be found in Appendix A, with corresponding listings in Appendix B.

The low cycle ΔK ranges included in the testing were 15, 20, 30 and 40 MPa \sqrt{m} . The low cycle hold times that included 2, 5, 10 and 180 seconds were expected to cover the regimes in low frequency loading in which the low cycle crack growth is time dominated (creep crack growth) and the regime in which the number of low frequency cycle influences crack growth rate (combination of creep and fatigue crack growth). The lower end of the low cycle hold period range (i.e., 2 and 5 seconds) was expected to show the effect of accumulated low frequency cycles on the low cycle crack growth rate. The series of data plots representing crack growth rates versus high cycle ΔK for constant low cycle ΔK and low cycle hold time obtained in this study show several interesting trends.

In the curves of crack growth versus high frequency ΔK distinct regimes can be seen. As shown in Figure 29, three types of behavior were observed over the range of low frequency ΔK and hold times investigated. In type 1, the crack growth rate versus high cycle ΔK remained relatively constant in the low cycle dominated regime prior to the rapid increase in crack growth rate in the high cycle dominated regime. Type 2 behavior was characterized by retardation of crack growth rate by the high frequency cycle in the low cycle dominated regime. Type 3 behavior was typical of the lowest low cycle ΔK studied, in which the low cycle ΔK was below the crack growth threshold and no crack growth could be measured in the low cycle dominated regime. In all these cases distinct low cycle and high cycle dominated regimes could be observed. However, the transition between these two regimes was not always distinct due to the retardation effect.

TABLE 4: COMBINED CYCLE TEST INCLUDING A 200 HZ HIGH CYCLE FREQUENCY

(All testing was conducted at 649°C (1200°F) and with a low cycle R ratio of 0.1)

		LOW CYCLE HOLD TIME (sec)			
		2	5	10	180
	15	TEST #'s 24, 25	21, 23	20	42, 43
Low Cycle Maximum	20	35, 37	30, 28	26, 27	39, 40
K (MPa 🗹 m)	30	46	31, 32	47	41
	40	36, 38	33, 34	48	44



FIGURE 29 Characteristics of the High/Low Frequency Interaction Showing the Three Types of Behavior Observed in This Study. The Points Correspond to testing with a Low Frequency Δ K of 20 MPa \sqrt{m} , a Low Cycle Time of 10 Seconds and a High Cycle Frequency of 200 Hz.

Figure 30 shows a curve corresponding to a low cycle ΔK of 15 MPa \sqrt{m} and a hold time of 5 seconds. The data in this representation corresponds to a test with increasing high frequency ΔK . Prior to data acquisition in the increasing high cycle ΔK mode, the crack was allowed to grow with a systematically decreasing ΔK until a crack growth rate on the order of 5 x 10^{-4} mm/sec (2 x 10^{-6} inches/sec) was achieved. This precaution was taken to eliminate the effects on crack growth of the prior precycling. The data presented in Figure 30 is characteristic of threshold fatigue crack growth data which generally exhibits increasing growth rate and decreasing slope with increasing ΔK when crack growth versus ΔK is plotted on log-log axes. The lower level of this curve corresponds to a growth rate of 1.3 x 10^{-8} inches per high frequency cycle (3.30 x 10^{-7} mm/cycle) which would definitely be in the threshold regime for Inconel 718.

Figure 31 shows the results of a test conducted with a low frequency ΔK of 20 MPa \sqrt{m} and a hold time of 5 seconds. It was carried out with a sequence of loads intended to illustrate an important aspect of the retardation effect that is very pronounced at a low frequency ΔK of 20 MPa \sqrt{m} . The line drawn through the experimental points has arrows drawn to show the sequence of points as they occurred during the test. The initial loading up to point A seems to give rise to a measurable retardation, and changing the high frequency load range to that at point B rapidly accelerates the retardation. This results in a more severe retardation in the 0.762mm (0.030) inches of growth beyond point B than was accomplished in the 2.79mm (0.110 inches) of growth with the high cycle ΔK range around point A. (Each point represents 0.010 inches, 0.254mm, of crack growth). Beyond point B the crack growth rate decreases rapidly, reaches a minimum value and then starts to increase. At point C just beyond the minimum value of crack growth, a lower high frequency load range was applied (the new level of high cycle ΔK is represented by point D). The crack growth rate increases from point D to E showing a gradual elimination of the retardation effect. At point E the load range was again increased to point F and crack growth continued in the high cycle dom nated regime.

As the low frequency ΔK increases, the retardation effect generally becomes less pronounced. The data for a low cycle ΔK of 30 MPa \checkmark m and a low cycle hold time of 5 seconds appears in Figure 32. While the high frequency load results in a factor of four reduction in crack growth rate for a low cycle ΔK of 20 MPa \checkmark m, at a low cycle ΔK of 30 MPa \checkmark m the reduction in crack growth rate is only a factor



FIGURE 30 Results of Combined High/Low Frequency Test With a Low Cycle ΔK of 15 MPa \sqrt{m} and a Low Cycle Hold Time of 5 seconds.



FIGURE 31 Results of Combined Cycle Test With a Low Frequency ΔK of 20 MPa \sqrt{m} and a Hold Time of 5 Seconds. The Line is Shown to Show the Sequency of Points.

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FIGURE 32 Results of a Combined Cycle Test With a Low Frequency ΔK of 30 MPa \sqrt{m} and a Hold Time of 5 Seconds.

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of 2. As shown in Figure 33, a low cycle ΔK of 40 MPa \sqrt{m} shows no measurable retardation associated with high frequency loading.

Figure 34 shows the effect of varying cycle time on the crack growth behavior with a low cycle ΔK of 15 MPa \sqrt{m} . No distinct trend is apparent and there is little deviation between these curves. Figure 35 shows the effect of cycle time ranging from 2 seconds to 180 seconds on the crack growth behavior with a low cycle ΔK of 20 MPa \sqrt{m} . The only significant feature in this group of tests is that with a 180 seconds hold time there appears to be a more severe retardation.

A comparison of crack growth rate versus high cycle ΔK for a hold time of 5 seconds and several values of low cycle ΔK appears in Figure 36. As expected the crack growth rate in the low cycle dominated regime increases as ΔK increases. A similar comparison is made in Figure 37 but with a low cycle hold time of 180 seconds and roughly the same behavior can be observed.

B. Results of Combined Cycle Tests with an 1800 to 2000 Hz High Cycle Component and Comparison with Lower Frequency Results

The parameters covered by the 1800 to 2000 Hz combined cycle testing are indicated in Table 5. Tests 60 through 66 were performed on specimens from a second heat of material. The crack growth rate in the low cycle dominated regime was quite different from the previous batch of material. The newer material has crack growth rates lower by a factor of 6 to 8 at some low cycle ΔK levels. Since, it is desirable to evaluate the effect of frequency without the complication of lot to lot material variation, fatigue crack growth testing near 2000 Hz was also performed on material from the older lot of material. Tests 67 through 75 represent tests from the same lot used for the 200 Hz tests. The complete set of data plots and listings for the 1800 to 2000 Hz combined cycle tests may be found in Appendices B and C respectively.

The dynamic tests performed on the laterally supported and damped specimen indicated that there is greater consistency in dynamic behavior at 1825 Hz than at 2000 Hz. There is also a greater high frequency load capability at 1825 Hz.



FIGURE 33 Results of a Combined Cycle Test with a Low Cycle ΔK of 40 MPa \sqrt{m} and a Hold Time of 5 Seconds.



FIGURE 34 Comparison of Crack Growth Rate Versus High Cycle ∆K for several Hold TImes With a Low Cycle ∆K of 15 MPa √m.



FIGURE 35 Comparison of Crack Growth Rate Varsus High Cycle ΔK for Several Hold Times and a Low Cycle ΔK of 20 MPa \sqrt{m} .



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FIGURE 36 Comparison of Crack Growth Rate Versus High Cycle ΔK for Several Low Cycle ΔK ranging From 15 to 40 MPa \sqrt{m} With a Low Cycle Hold Time of 5 Seconds.


FIGURE 37 Comparison of Crack Growth Rate Versus HIgh Cycle ΔK for Several Low Cycle ΔK Ranging From 5 to 40 MPa \sqrt{m} With a Low Cycle Hold Time of 180 Seconds.

TABLE 5: COMBINED CYCLE TEST INCLUDING AN 1800 to2000 Hz HIGH FREQUENCY LOAD COMPLETED TO DATE





FIGURE 38 Comparison of Results for 200 and 1325 Hz for a Hold Time of 5 Seconds and a Low Cycle ΔK of 30 MPa \sqrt{m} .

Tests 60 through 66 in this higher frequency series were conducted with several high cycle frequencies in the range of 1800 to 2000 Hz. The advantages of testing at 1825 Hz became apparent and, therefore, beyond test 66, 1825 Hz was used as the high cycle frequency. Comparison of results for 200 Hz and 1825 Hz tests may be found in Figures 38 and 39 which present data for a 5 second hold time and for a ΔK of 30 MPa \checkmark m and 20 MPa \checkmark m respectively. A feature that the 1825 Hz tests show in these figures is a less pronounced retardation than at 200 Hz. Other tests performed near 2000 Hz show similar results. In Figure 39 for a low cycle ΔK of 20 MPa \checkmark m, on the onset of high cycle activity appears to occur at a lower high frequency ΔK at 1825 Hz than at 200 Hz. A distinct low cycle dominated range of high frequency ΔK is apparent at both frequencies.

A comparison between combined cycle loading with a high frequency component of 200 Hz and 10 Hz is presented in Figure 40. The 10 Hz data is from Reference 1. The apparent onset of high cycle behavior is about the same. The initial slopes of the high cycle dominated regime are significantly different with the 200 Hz data having a larger slope. This behavior would be expected in the high cycle dominated regime in which the number of high frequency cycles determines the rate of crack growth.



FIGURE 39 Comparison of Results for 200 and 1825 Hz for a Hold Time of 5 Seconds and a Low Cycle ΔK of 20 MPa \sqrt{m} .



FIGURE 40 Commarison of Results for 10 and 200 Hz for a Hold Time of 180 Seconds and a Low Cycle AK of 30 MPa 4m.

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IV EVALUATION OF MECHANISMS AND MODELLING ASSOCIATED WITH FREQUENCY EFFECTS AND COMBINED HIGH/LOW CYCLE INTERACTION

There have been several studies of frequency effects in nickel base and other alloys up to frequencies of 20,000 Hz. Investigation of combined high and low frequency interactions in fatigue have also been performed. In this section the important observations and conclusions from these studies will be summarized. The test results from this program will then be discussed in the context of these previous studies.

A. Background

References 8-11 provide a review of mechanisms that apply to fatigue crack growth of nickel base alloys at elevated temperatures. These papers deal with both the initiation and propagation of fatigue cracks and the influence of frequency on these processes. Frequency effects are evaluated in terms of the effect of frequency on "slip character", which is the degree to which dislocations disperse during plastic deformation. The two extremes in slip character that nickel base alloys have exhibited are planar slip and wavy or homogeneous Planar slip is characterised by the concentration of dislocations in slip. planar arrays with planar shear offsets produced on polished surfaces transverse to the crack plane and parallel to the direction of propagation. This type of slip and its associated deformation is favored by low stacking fault energy, ordering, the presence of coherent precipitates, low temperatures, and small strains. Austenitic stainless steel and nickel base alloys both exhibit planar slip at ambient temperatures. Wavy or homogeneous slip on the other hand, is characterized by uniformly distributed, nonplanar dislocation arrangements with an associated rumpling of the surface transverse to the crack plane and parallel to the growth direction. Wavy slip is favored by high stacking fault energies, incoherent precipitates or particles, large strains, and elevated temperatures. Most metals including stainless including stainless steels and nickel base alloys exhibit wavy slip at temperatures greater than $0.4T_{M}$ (T_M = melting temperature) because a thermally activated process allows dislocations to cross slip and climb out of their original slip planes. Wavy slip can occur in both transgranular and intergranular fracture modes. The fact that wavy slip occurs by a time dependent, thermally activated process in iron and nickel base alloys at elevated temperatures has significant impact on the frequency dependence of

fatigue. As the frequency or strain rate increases the degree of slip dispersal decreases, i.e., when the characteristic time constant associated with slip dispersal becomes larger in relation to the cycle time associated with deformation, slip becomes more concentrated on certain planes. It has in fact been observed that as cycling frequency increases, slip becomes similar to that observed at ambient temperatures in nickel base alloys. At higher frequencies as with lower temperatures, planar slip tends to dominate.

Fatigue life over the broad range of frequency from .033 Hz to 1000 Hz for Udimet 700 at 1400°F (760°C) is presented in References 8 and 9 and shown in Figure 41. From .033 Hz to 10 Hz, fatigue life at the given strain range increases by a factor of 100. Over this frequency range there are changes in the site of crack initiation. At the lowest frequency initiation occurs at surface connected grain boundaries and the initial mode of fracture in intergranular. As frequency increases in this range of frequency, intergranular cracking generally associated with creep and oxidation become less dominant giving way to transgranular fracture. At a frequency of 3 Hz, the fracture is almost entirely transgranular. With an increase in frequency from 10 to 1000 Hz, fatigue life is reduced by a factor of seven because of the concentration of deformation in fewer slip bands and the resulting accelerated crack initiation and propagation. Reference 9 suggests that the main reason for the reduced fatigue life beyond a frequency of 10 Hz may be associated primarily with the number of cycles required for crack initiation.

The nature of crack initiation has been shown to change with changing frequency. Stage I and Stage II are two classifications of fatigue crack initiation. Stage I crack initiation is favored by low temperatures and high frequencies, i.e., the same conditions that lead to planar slip. Low frequencies and high temperature on the other hand, favor Stage II crack initiation. Additional observations regarding the influence of frequency on crack initiation in nickel base alloys is given in Reference 10. Fatigue cracks in Udimet 700 at 1400°F (760°C) are shown to initiate in an intergranular mode from a surface intitation site at frequencies of 0.033 to 0.33 Hz. The crack then extends intergranularly along the surface to a depth of 1 to 3 grain 'iameters below the surface and then changes to a transgranular Stage II mode. At frequencies in the range 3 to 1000 Hz, crack propagation began in the Stage I transgranular mode and changed to a Stage II mode. It was also observed over the many specimens examined that low



Figure 41 The Effect of Frequency on the Number of cycles and Time to Failure of **V**-700 at 1400°F (760°C) and a Stress Range of 85 Ksi. (9)

trequencies favor surface intergranular crack initiation and intergranular crack propagation. High frequencies on the other hand favor subsurface initiation at grain boundaries or twin boundary intersections and transgranular crack propagation.

A study of the influence of cyclic frequency on the fatigue properties of single crystal MAR-M-200⁽¹¹⁾ showed results similar to those for Udimet 700. Testing was performed on MAR-M-200 at frequencies from 0.033 Hz to 1030 Hz over the temperature range 1400°F (760°C) to 1800°F (982°C). The number of cycles to failure at 1400°F (760°C) and 1550°F (787°C) reached a peak in the range of 1 to 10 Hz. Stage I crack initiation was favored at the lower temperatures and higher trequencies and Stage II crack initiation at the higher temperatures and lower frequencies. At 1030 Hz crack initiation and propagation occurred entirely in the Stage I fracture varied according to 110 slip planes. Generally, the amount of Stage I fracture varied according to temperature and frequency as shown in Figure 42. The nature of the fracture was attributed to degree of slip homogeneity. Stage I is favored by inhomegeous planar slip and Stage II is tavored by homogeneous slip. In almost all specimens of MAR-M-200 cracks initiated at subsurface micropores.

Clavel and Pineau⁽¹²⁾ studied the effects of frequency and wave form on the fatigue crack growth of Alloy 718 (a nickel base superalloy) in the frequency range between 5×10^{-3} Hz and 20 Hz at 298°K and 823°K. The variation in fatigue crack growth rate with frequency that they observed at 823°K is summarized in Figure 43. Consistent with the observations on Udimet 700, Nimonic 90, and MAR-M-200 single crystals, the fatigue crack growth rate decreases with increasing frequency in the regime in which environmental and time dependent material deformation processes creep effects can operate. Fractography revealed that this decrease in fatigue crack growth rate (FCGR) is accompanied by a change in fracture mode from intergranular to transgranular. Suprisingly, they observed through TEM examination of the substructure that higher strain rates promoted more homogeneous plastic deformation while low strain rates favor inhomogeneous deformation and the formation of twins. The crystallographic aspects of the fracture surface observed at room temperature in the threshold regime is attributed to the decohesion along the twin deformation bands.



FIGURE 42 The Percentage of Stage I Fracture in the Fatigue Zone as a Function of Cyclic Frequency at Temperatures of 1033, 1116, 1200, and 1255 K. (9)



FIGURE 43 Variation of FCG Rate (da/dN) With Stress Intensity Factor (△K) and Frequency (√) at 823 K (Sinusoidal Load) for Inconel 718. (12)

A paper by Sullivan et. al.⁽¹³⁾ discusses the effect of cycling frequency in the very low frequency regime for the nickel base superalloys Udimet 700 and MAR-M-200. Creep tests were performed on these materials at $955^{\circ}C$ in air with periodic unloading. The time intervals between unloadings were on the order of 15 minutes to 5 hours. The main observation made regarding the effects of unloading is that it produced accelerated creep rate in both Udimet 700 and directionally solified MAR-M-200.

Scarlin⁽¹⁴⁾ also studied the effect of frequency in the range 10^{-4} to 10^2 Hz on the fatigue crack growth of two nickel base superalloys: Nimonic 105 at 750°C and IN 738 LC at 850°C. The results show the expected decreasing crack growth rate with increasing frequency.

The influence of environment on the frequency dependence of fatigue has also been investigated. For example, in the lower frequency regime (up to 1.7 Hz) Solomon and Coffin⁽¹⁵⁾ studied the effect of frequency on the fatigue crack growth of A286 at 1100° F in both air and vacuum. They observed generally that the crack growth mode and frequency dependence of crack growth rate varied with frequency in the manner shown in Figure 44. This representation of the crack growth data shows that specimens tested in air and vacuum both have frequency regimes of intergranular and transgranular fracture but with different behavior in the lower cycle regime. Likewise, both air and vacuum tested specimens have a frequency above which the crack growth rate is independent of frequency. This study also shows that the dependence of crack growth rate (in growth per cycle) may be represented as follows.

$$\frac{\mathrm{d}C}{\mathrm{d}N} = \phi C (\Delta \varepsilon_{\rho})^{\alpha} v^{k-1} \qquad (1)$$

where $\Delta \varepsilon_{\rho}$ is the plastic strain range for the specimen used in their experiments, C is the measured crack length, dC/dN is the crack growth rate and ϕ , α , and k are constants. Each regime shown in Figure 44 is characterized by a different value of k; the pure cycle dependent regime has a k value of 1.0.

The tests performed in vacuum generally have a lower growth rate than in air. The difference in fatigue crack growth rate, however, decreases with increasing frequency and the results converge at high frequency. These results suggest



FIGURE 44 Schematic Comparison of the Air and Vacuum Crack Growth Behavior.

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that as frequency increase the effect of environment reduces, or in effect the crack at higher frequencies out runs the processes of oxidation that degrade the tatigue crack growth properties.

Another study demonstrating the influence of environment on frequency effects in the fatigue crack growth involves 200 Maraging steel in a salt water environment and is reported in Reference 16. A significant frequency effect on the crack growth of this alloy in salt water in the frequency range 0.17 Hz and 3.3 Hz. At 3.3 Hz, it was found that the salt water solution had little effect on the crack growth rate as compared to the results in air. There was a factor of ten increase in crack growth rate when the frequency was reduced to 0.017 Hz. This along with the comparisons for different gaseous environments demonstrates that environment may be responsible for much of the frequency effect.

An important aspect of frequency effects in component life prediction is the effect of frequency on the threshold stress intensity factor range $(\Delta K_{\rm th})$. There are reports on the aspect for several materials. Mautz and Weiss⁽¹⁷⁾ reported the effects of frequency on $\Delta K_{\rm th}$ for D6ac steel at room temperature for both air and argon environments. No frequency effects on threshold behavior were observed for an air environment between frequencies of 100 and 375 Hz. In dry argon, however, the results for 100 Hz were slightly higher than those at 375 Hz.

A very extensive study of fatigue crack growth properties of titanium alloys used in aircraft engine compressors was performed by Beyer, Sims and Wallace⁽³⁾. Frequency effects up to 1000 Hz on the fatigue crack growth properties of Ti-6Al-2Sn-4Zr-6Mo, Ti-8A&-1Mo-1v, and Ti- 6Al-2Sn-4Zr-2Mo were investigated at room, 600°F, 800°F, 900°F and 1000°F for several R ratios for crack growth rates down to the threshold regime. For the higher R ratios such as 0.5 and 0.7 there is a considerable reduction for all three alloys when the frequency is increased to 1000 Hz from 0.17 Hz at elevated temperatures. The threshold stress intensity factor likewise reduced on increasing the frequency to 1000 Hz. The variation in crack growth in the frequency range 0.017 Hz to 30 Hz was much less than that between 30 and 1000 Hz.

The highest test frequency in fatigue testing that we were able to find in the literature was that used by St. Stanzl and Mitsche⁽¹⁸⁾ who performed crack

growth tests on 0.04%C steel, chromium steel 20A13 (0.29.C, 13%a), and pure molybdenum at 20 kHz. They conclude that their results in terms of crack growth rate versus ΔK are similar to those for 10 Hz provided by another investigation.

Combined high cycle/low cycle loading has been investigated for several materials. The frequencies for the high cycle and low cycle components represented in these studies cover a very broad range in both loading components. At the lowest extreme in low cycle loading there is the low cycle frequency of zero with the high cycle frequency in the range that will with sufficient amplitude cause fatigue crack growth. This combined cycle interaction is often referred to as creep-fatigue interaction. (19-24) Another group of papers and reports (1, 25-28) deal with high cycle/low cycle interaction where the high and low cycle components correspond to those that are encountered in rotating machinery. The low cycle component has a cycle period on the order of seconds to several hundred seconds and the high cycle frequency ranges from 10 Hz to several thousand Hz.

Several studies have shown that load cycling can have an effect on the creep rate. Both increases and decrease. in creep rate have been observed when cycling is applied. The softening has been attributed to and increased mobility of piled up dislocations as a result of the fatigue cycling assisting the dislocations to overcome obstacles and "friction" stress fields in the slip plane. The hardening effect has been explained in terms of migration of solute atoms or dispersed point defects towards free dislocations. Venkiteswaran et. al.⁽¹⁹⁾ who studied the precipitation hardened alloy Inconel Alloy X-750 attributed the reduction in creep rate due to an applied fatigue cycle to the formation of complex dislocation tangles and vacancy condensation along dislocation lines. A change in fracture mode from intergranular to transgranular was also observed with the application of the 555 to 910 Hz fatigue loading.

Atanmo and McEvily⁽²⁴⁾ reported on the creep-fatigue interaction during crack growth of aluminum alloy 5052 at 400°F. Conducting tests with ramped loading and hold times ranging from 30 to 65 seconds, as well as with steady load, theyobserved that cyclic-creep lifetimes can exceed creep lifetimes, perhaps as a result of the reversal of the creep process at the crack tip during the off-load period of the test. There are several investigations of high cycle/low cycle interaction motivated by design considerations in rotating machinery such as generating plants, gas turbines and compressors. These studies all involve a low cycle loading component consisting of a trapezoidal waveform with a high frequency component applied during the upper level hold time as shown in Figure 45. Included in these studies are those performed at Portsmouth Polytechnic Institute on Ti-6Al-4V.⁽²⁵⁻²⁷⁾ The dwell period for this test was 6.8 seconds. Testing was performed at room temperature with a high cycle frequency of 150 Hz. Fatigue crack propagation experiments with increasing load and high cycle load levels were undertaken using minor to major amplitude ratios (Q) of 0, 0.1, 0.2 and 0.3. Figure 46 shows the effect of amplitude ratio (Q) on the measured FCG rates.

A series of tests were conducted to determine the value of high frequency ΔK under major-minor cycling corresponding to measureable influence of the high cycle component on crack growth. A step down procedure was used to determine the threshold for high cycle activity. Table 6 lists the conditions for the onset of minor cycle damage and onset of fast fracture. The authors also evaluated the appropriate manner of predicting crack growth rate under combined cycle loading. The two approaches to crack growth prediction evaluated by the authors are the linear summation of the major and minor cycle crack growth rates measured individually and the representation of the complex loading in terms of its RMS value. A comparison of the experimental results with the predicted results are shown in Figures 47 and 48. In these cases, the crack growth is dominated by the high cycle (minor cycle) loading and the prediction of both the linear summation and RMS representation are satisfactory.

Goodman and Brown⁽¹⁾ report on several combined cycle tests on alloy 718 at 649° C with a high cycld frequency of 10 Hz and the same loading profile as used in the studies of References 25 and the present study. Features similar to those found in the present study at 200 and 1825 Hz were observed including retardation in the low cycle dominated regime and the existence distinct high and low cycle dominated regimes. Also included in the program conducted by Goodman and Brown, tests with a high cycle frequency of 100 and 200 Hz.



FIGURE 45 a - Major Cycles Only; b - Major and Minor Cycles (Minor/Major Amplitude Ratio $Q = \Delta K_{minor} / \Delta K_{major}$)



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FIGURE 46 Effect of Amplitude Ratio on FCG Rates for Major and Minor Cycles. (25)

Ampli	R	Onset of minor cycle activity			Onset of fast fracture	
tude ratio		ΔK_{minor}	ΔK _{major}			
0.02	0.982	1,5	75.0	75.8	1.3	63.1
0.04	0.965	1.6	40.0	40.8	2.5	63.1
0.1	0.914	1.7	17.0	17,9	6.0	63.3
0.2	0.835	2.1	10.5	11.6	11.6	63.6
0.3	0.762	2.3	7.7	8.8	16.7	63,9

TABLE 6 Conditions for the Onset of Minor Cycle Damage and Onset of Fast Fracture (ΔK Values in MPa \sqrt{m}).



FIGURE 47 Linear Summation of FCG Rates (Damage: A-Associated With Applied Major Cycle: B-Associated With Applied Minor Cycles: C-Given by Summation of Major and Minor Cycle Damage). (25)

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FIGURE 48 Analysis of Major-Minor Fatigue Crack Growth Rates in Terms of $\Delta \kappa_{RMS}$ (25)

B. Evaluation of Fatigue Crack Growth Mechanisms Under Combined Cycle Loading

The study of Venkiteswaran et. al., $^{(19)}$ reports the results of creep testing with a superimposed small vibratory stress on the axial creep behavior of a high temperature nickel base alloy, Inconel X-750. This work demonstrated that the creep rate was lower and rupture life higher by an order of magnitude when a 500 to 900 Hz vibratory stress was applied transverse to the axial creep load. This effect was attributed to the formation of complex dislocation tangles, vacancy condensation along dislocation lines and crack tips and also a change in fracture mode from purely intergranular fracture to a mixture of intergranular, fatigue and cleavage modes. It was suggested that the application of the high frequency loading, therefore, made creep crack propagation more difficult along the matrix containing γ' precipitates. Since the heat treated Inconel 718 used in this study likewise contains γ' (Ni3 A2-Ti) as well as γ'' (Ni3Cb) precipitates, this mechanism could apply in the present study. The changing mode of fracture observed by them is consistent with the fractographic features of the combined cycle crack growth specimen that we investigated.

Fractographic examination was performed on specimens subjected to 200 Hz cyclic load in order to obtain information regarding fracture mechanisms. Areas on a 200 Hz specimen were examined by Scanning Electron Microscopy (SEM) and Scanning Transmission Electron Microscopy (STEM). All of these photographs correspond to specimen 28. The fracture surface includes areas corresponding to creep crack growth with no high frequency loading and areas corresponding to combined high/low cycle loading with crack growth both in the low and high cycle dominated regimes. The fracture surfaces of these regions show distinct differences. Figures 49 and 50 show SEM photomicrographs of the purely low cycle and combined cycle regions respectively. The purely low cycle region shows intergranular fracture typical of creep crack growth. With the application of a high frequency load range at a level that maintained the low cycle dominated behavior, the fracture becomes predominantly transgranular with the appearance of fatigue striations.

Replicas were taken of the fracture surface and subsequently shadowed with chromium and coated with a film of carbon. These replicas were then examined in an SEM with a transmitted beam. The resulting photomicrographs for a region of low cycle loading only, are shown in Figure 51. The intergranular nature of the









1000x

E5636

FIGURE 50: Scanning Electron Microscope (SEM) photomicrograph of a region in which combined cycle loading (with 200 Hz high cycle load) was applied. fracture is clearly shown in these photomicrographs. In Figure 52 are shown the STEM photomicrographs for a region which had experienced combined cycle loading. The striation pattern is this predominantly transgranular fracture seems to show grouping of striations. At the higher magnification of 10,000x the pattern appears to be obscured, probably by oxidation at the test temperature of $649^{\circ}C$.

Additional STEM photomicrographs were made on a specimen tested with a high cycle frequency of 1825 Hz (specimen 67). Without high frequency cycles applied, the fracture surface showed the expected intergranular fracture. Figure 53 shows the fracture surface in the low cycle dominated regime where the high cycle ΔK is large enough to cause retardation. A striation pattern is apparent. Figure 54 shows the fracture surface well into the high cycle dominated regime. The striation pattern in this region is more pronounced and shows a greater spacing corresponding to the increased crack growth.

The relationship between fatigue crack growth and high cycle ΔK for constant low cycle ΔK show three regimes. At the lower limit of $\Delta K_{\rm HC}$ the low cycle loading dominates the rate of fatigue crack growth. In an intermediate range of $\Delta K_{\rm HC}$, the high cycle loading causes a retardation of the crack growth rate. At the highest values of $\Delta K_{\rm HC}$, crack growth rate is dominated by the high cycle loading with crack growth determined by the number of high frequency cycles. The low and high cycle dominated regimes are distinct but the transition between the two regimes is obscured by the retardation effect. The behavior of alloy 718 at 649°C revealed by this study is similar to that shown by Goodman and Brown⁽¹⁾ who studied the interactive effect of this alloy at 649°C with a high cycle dominated regimes as well as a regime of $\Delta K_{\rm HC}$ where retardation occurred were also apparent. The investigation of Powell et. al.,⁽²⁵⁾ on Ti-6-4 showed regimes of $\Delta K_{\rm HC}$ where the high cycle loading was either active or inactive, but a retardation effect was not apparent.

The retardation effect was unexpected and an experiment was carried to gain insight into its origin and characteristics. The experiment summarized in Figure 31 shows the rate (with respect to crack length) at which the retardation effect develops and also the rate at which it relaxes. There seems to be a crack growth interval of about 1mm (0.0394 inches) required for the retardation effect





FIGURE 51: Scanning Transmission Electron Microscope (STEM) photomicrographs of a region in which only low cycle loading was applied.



Figure 51 (Cont'd) Scanning Transmission Electron Microscope (STEM) photomicrographs of a region in which only low cycle loading was applied.









FIGURE 52: Scanning Transmission Electron Microscope (STEM) photomicrographs of a region in which combined cycle loading (with 200 Hz high cycle load) was applied.

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Figure 52 (Cont'd) Scanning Transmission Electron Microscope (STEM) photomicrographs of a region in combined cycle loading (with 200 Hz high cycle load) was applied.



E5894

x3000

E5893

x10,000

Figure 53 STEM Photomicrographs of a Region on the Fracture Surface of Specimen #67 Corresponding to the Low Cycle Dominated Regime where the High Cycle K is Large Enough to Cause Retardation

Α.

в.



E5785

Α.

x3000

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E5786

x10,000

Figure 54 STEM Photomicrograph of a Region on the Specimen #67 Fracture Surface Where the High Cycle Component Dominates Crack Growth. to subside. The plastic zone size associated with the crack, however, is a small function of this length.

The size of the plastically deformed region (R) ahead of the crack as given by the Dugdale model (28) neglecting the effects of creep is:

$$R = \left\{ \sec[1/2\pi(\sigma/\sigma y)] - 1 \right\} a$$

where σ is the applied stress, σy is the yield strength and a is the half crack length.

As calculated using this expression at the crack length corresponding to the retardation relaxation in Figure 31 and assuming a yield strength 980 MN/m^2 (140 ksi) the plastic zone size is 0.20mm (0.008 inches). A possible explanation of the fact that the affected region is considerably larger than the calculated plastic zone length is that creep stress relaxation results in a larger characteristic zone where structural changes important to retardation effects occur. Reference 29 demonstrates that crack tip stresses can be modified significantly by creep. The most significant influence of creep relaxation as shown by Reference 29 is the reduction of the stress gradient beyond the crack tip, i.e., the development of a more uniform distribution of stress in a region that includes the above calculated "plastic zone" and an area further from the crack tip. However, it is unlikely that creep can have such a pronounced influence on the crack stress distribution.

An alternative explanation of the long relaxation interval is suggested by the study of Venkiteswaran⁽¹⁹⁾ that showed that the high frequency loading affects the creep rate versus stress constitutive properties. The modification of the creep rupture processes may in turn modify the residual plastic deformation remaining in the wake of the advancing crack (crack closure).⁽³⁰⁾ Such a concept would allow the possibility of the effect persisting well beyond the above calculated plastic zone without postulating a significant modification in the crack tip stresses due to creep relaxation effects. The crack growth interval of four or five times the plastic zone size is in fact characteristic of the development of closure effects.

cycle frequency is applied. However, when the retardation effect occurs with this lower high cycle frequency, a longer time period is required to reach the minimum crack growth rate. This would be expected if the retardation effect is related to the accumulated number of high frequency cycles.

A feature of the high cycle loading revealed by this and other studies is that there is a sharply defined value of transition ΔK (ΔK_{tr}) associated with the dominance of ΔK_{HC} for crack growth under combined cycle loading. For the range of conditions investigated, the influence of high cycle loading on crack growth below ΔK_{tr} can essentially be ignored. This is consistent with the observations of Powell et. al.⁽²⁵⁾ who performed combined cycle crack growth experiments on Titanium - 6 - 4 at ambient temperatures with a high cycle frequency of 150 Hz and with all of the combined cycle testing results of Goodman and Brown⁽¹⁾ performed on Inconel 718 at 649°C for a high cycle frequency of 10 Hz. Furthermore, there is little variation of this transition ΔK with frequency distinguishable beyond the ΔK_{tr} variation intrinsic to the material.

Considering the various features of the crack growth rate beyond ΔK_{tr} , i.e., that growth rate depends on number of cycles, that it is sharply defined by a threshold value, and that is shows a relationship between growth rate and ΔK_{HC} similar to that for stage I crack growth leads to the conclusion that it could be represented by a relationship of the form:

 $\frac{da}{dN} = c \left(\Delta \kappa_{HC} - \Delta \kappa_{tr}\right)^{m}$

where C is a constant and da/dN is crack growth rate in terms of crack extension per high frequency cycle. This relationship has been used to describe crack growth in the threshold regime (stage I) with a constant R ratio (Kmin/_{Kmax}). The R ratio in the high cycle dominated regimes in the experiment conducted in this study, varies a small amount since they correspond to increasing ΔK_{HC} and constant ΔK_{LC} tests. The ΔK_{tr} is the above expression is expected to vary with low cycle ΔK and perhaps hold time but as shown by the present investigation it is essentially invariant with respect to frequency. This feature is helpful to the design since an acceptable level of high cycle loading can be established without concern for the frequency of the superimposed high frequency load. is essentially invariant with respect to frequency. This feature is helpful to the design since an acceptable level of high cycle loading can be established without concern for the frequency of the superimposed high frequency load.

The investigation of Cell and Leverant⁽⁸⁾ showed a pronounced influence of frequency on fatigue life of nickel base alloys in the frquency range of 10 to 1000 Hz. In this range they observed that fatigue life decreases with increasing frequency. Comparing the results of the present investigation with those of Goodman and Brown,⁽¹⁾ there is little variation in ΔK_{tr} over the frequency range to 2000 Hz. This fact and the fact that the crack growth rate per cycle versus ΔK_{HC} beyond ΔK_{tr} does not increase with increasing frequency leads to the conclusion that the decreasing fatigue life with increasing frequency observed by Gell and Leverant⁽⁸⁾ is associated primarily with crack initiation.

C. Consideration of High/Low Cycle Interactions in Crack Growth Life Prediction of Engine Systems

Attention has been devoted recently to the effects of gas turbine engine load spectra on crack propagation. This is a result of increased performance requirements for U.S. Air Force gas turbines and the resulting high operating stresses and severe service environments experienced by gas turbine components. Many of the investigations are associated with the development of the advanced life management concept and focus on engine disks.

An important aspect of life prediction under engine loading spectra is the interaction of the low and high cycle components in crack growth of turbine disks. The cycle period associated with the low frequency cycle (low cycle) loading is on the order of seconds to several hundred seconds. A wide range of loading rates and load levels may also be involved in the low cycle loading. The high frequency cycle (high cycle) loading would typically involve frequencies on the order of hundreds to several thousand hertz. Important to accurate life prediction is establishing the manner in which each of these features of the engine disk loading profile contribute to crack growth and how these features interact. The specific aspects of combined cycle loading that must be addressed are the following:

- Establishment of the limits of high cycle loading under which the disk can be safely operated.
- How cumulative damage rules should be applied when combined high cycle/low cycle loading contribute to crack growth.
- The degree to which the high cycle and low cycle loading influence each others contribution to crack growth.

There are previous studies of load spectrum interaction in crack growth of aircraft engine components that deal with periodic overloads, overload/underload combinations and periods of sustained load interspersed with relatively constant amplitude loading. An example of such a study in that of Macha et. al.⁽³¹⁾ which considered these effects on IN-100 and evaluates the applicability of crack growth rate models for engine complex loading spectra. Another study that addresses the effects of flight loading in military gas turbine operation on the fatiuge crack growth of IN-100 and Waspaloy is summarized in References 32 and 33. This study addresses the effect of overload ratio and the effect of the number of cycles between overloads.

The simplest approach to crack growth prediction is a linear summation of crack growth on a cycle by cycle basis for the given loading profile. However, this approach has been shown to be inadequate for many situations of variable amplitude loading where retardation or acceleration can result from certain sequences of loading. Various approaches have been established to account for these effects including models based on crack closure⁽³⁴⁾ and the interactions in the crack yield zone⁽³⁵⁾.

In the present study, the applicability of a linear summation of high and low cycle crack growth contribution in predicting combined high/low cycle crack growth was investigated for Alloy 718 with a high cycle component of 200 and 1825 Hz. Figures 55 and 56 show a comparison between a combined cycle test result and a linear summation of crack growth rate calculated from crack growth data for the low and high cycle contributions measured individually. The manner of summing the individual high and low cycle components is shown schematically in these figures. The individual contributions were measured in an experiment with increasing $\Delta K_{\rm HC}$ superimposed on steady (not cycled) $\Delta K_{\rm LC}$ and in an experiment with ' pure low cycle loading with a triangular waveform and an R ratio of
0.1. For a high cycle frequency of 200 Hz, a low cycle ΔK of 30 MPa \sqrt{m} and a hold time of 180 seconds, Figure 55 shows a reasonable correspondence between actual results and those predicted from a linear summation in the high cycle dominated regime only. For the case of a high cycle frequency of 1825 Hz on the other hand there appears to be deviation in the high cycle dominated regime associated with a difference in ΔK_{tr} and a fair correspondence of crack growth rate in the low cycle dominated regime. These trends, however, are not necessarily representative. In this study, as well as that of Goodman and and Brown⁽¹⁾ a substantial variation in low cycle crack growth rate was apparent for tests carried out under identical conditions. Likewise, for given values of ΔK_{LC} hold time and frequency, a variation in ΔK_{tr} of 20% was apparent. The results of linear summation show a deviation from the combined cycle data that is in the range of variation in crack growth rate behavior for a given set of combined cycle parameters.

With some qualifications, a linear summation provides an adequate representation of combined cycle crack growth rate. In applying the linear summation approach to design, one must be aware of the fact that the low cycle crack growth rate, the retardation behavior and ΔK_{tr} can vary. An appropriate design curve to account for combined crack growth rate is the dashed line construction of Figure 55. The upper bound on low cycle crack growth is the horizontal dashed line. The upper bound on high cycle dominated crack growth is represented by the dashed line on the right side of the diagram. Together, the two curves define an upper bound on crack growth rate in the low cycle dominated, the retardation, and the high cycle dominated regimes. The crack growth rate predicted in the retardation regime would be a significant over estimate. However, this is necessary because the extent of retardation is not easily predicted and its benefit should, therefore, be ignored. Another important factor that must be kept in mind in applying a linear summation rule is that the relationship between crack growth rate and high cycle loading with a high level of mean load (i.e., large enough to cause creep crack growth) is not necessarily unique. For nickel base alloys, crack growth resulting from steady loads has been shown to exhibit non equilibrium behavior. In the present study, this effect was apparent when a small high cycle component was superimposed on a large steady load.



Figure 55 Comparison of Data (Points) with Growth Rate Predicted (Line) From a Linear Summation of Uncycled 1825 Hz High Cycle Data and Pure Low Cycle Data



Figure 56 Comparison of Data (Points) With Growth Rate Predicted (Line) From a Linear Summation of Uncycled 1825 Hz High Cycle Data and Pure Low Cycle Data

V CONCLUSIONS

The crack growth rate for Alloy 718 at 649° C was measured for combined cycle loading over ranges of low cycle ΔK , high cycle ΔK , low cycle hold time, and high cycle frequency. Several interesting trends in combined cycle crack growth were revealed. Generally, at lower values of high cycle ΔK , crack growth is dominated by the low cycle components of the load spectrum and with a sufficient level of high cycle loading, crack growth is determined predominantly by the accumulated number of high frequency cycles. These two features of combined loading crack growth behavior were consistent and straight forward. The nature of the transition from the low cycle to high cycle regime, however, depends significantly on the value of low cycle ΔK and to a lesser extent on high cycle frequency and low cycle hold time. The transition from low to high cycle dominated crack growth is obscured by a retardation effect.

This interactive effect is apparent under combined cycle loading generally at low values of ΔK and values of $\Delta K_{\rm HC}$ in a range between the purely low and high cycle dominated regimes. Fractographic examination reveals that it is associated with a change in crack growth mechanism from one characterized by intergranular fracture for pure low cycle loading to transgranular fracture for combined cycle loading. The degree of crack growth retardation appears to decrease with increasing low cycle ΔK and also decreases with increasing high cycle frequency. Another interesting featrue associated with the retardation effect is that a crack growth interval of several plastic zone sizes is required for its development or relaxation. Considering this fact and the fact that it becomes increasingly less pronounced with increasing $\Delta K_{\rm LC}$, leads to the conclusion that the retardation effect is associated with a change in the extent of plastically deformed material left in the wake of the advancing crack when high cycle loading is applied.

The slope of the log (crack growth rate per unit time) versus log ΔK_{HC} curve beyond ΔK_{tr} increases with increasing frequency as one would expect for a situation where crack growth rate in the high cycle regime is dependent on the number of cycles. The shape of the curve in the high cycle dominated regime is similar to that experienced for near threshold behavior (Stage I) observed in constant R ratio tests. (The R ratio of the high cycle loading in these studies varies with ΔK_{HC}). A crack growth rate prediction based on the linearly summed contributions for the high and low cycle components of loading correlate well in the case of a high cycle frequency of 200 Hz with some deviation in the low cycle dominated regime. For the case of 1825 Hz some deviation was observed for the high cycle dominated regime. This may be associated with the intrinsic variation in ΔK_{tr} .

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APPENDIX A

PERFORMANCE OF HIGH FREQUENCY SERVO-HYDRAULIC SYSTEM

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Figure A-2 Front Mount Type Constant Input-Frequency Response

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Front Mount Type Constant Input-Frequency Figure A-3 Response



Figure A-4 MTS System Fatigue Performance

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Figure A-5 Flow Versus Frequency for Akashi 37 gpm servo-valve

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APPENDIX B

DATA PLOTS FOR ALL TESTS

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APPENDIX C

DATA LISTINGS FOR ALL EXPERIMENTS

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RELULTS OF TEST NO. 1

Constant load test with a trapezoidal waveform having a fill second hold time, no high cycle loading, R = 0,1

Growth Rate (in./sec x 10 ⁶)	0.00715	0.1986	U.3024 4.687	8.404	9.927	12.37	13.58	13.20	11.57	7.285
Low Frequency (psi /in.)	17220.6094 18510 7766	27527.7383	26116.6055	26547.0898	26813.5547	27549.0000	28394.6094	29625.5273	30696.8203	28607.9453
Crack Length (in.)	0.1946	0.2183	0.2320	0.2387	0.2429	0.2345	6/97-0	0.2875	0.3045	0.3168
Time_5 (Sec x 10 ⁻⁵)	1.321 2.699	3.947 3.955	4.012	4.028	4.0.4	2 4 0 4 1 7 4 0 4 1			8/0.4	4.089

TABLE C.2

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RESULTS OF TEST NO. 2

Constant load test with a trapezoidal waveform having a 60 second hold time, no high cycle loading, R = 0.1

Growth Rate (in./sec)	0.00000125385 0.00000022738 0.00000075387 0.00000068074 0.00000068074 0.00000066424 0.0000006426
Low Freuguency AK (psi /in.)	16136.2617 16387.5547 16529.8945 16309.8945 16300.0469 16375.2969
Crack Length (in.)	0.1945 0.2000 0.2032 0.2086 0.2086 0.2086 0.2097
Time (sec)	2340-0000 7500-0000 11880-0000 19740-0000 20640-0000 22380-0000

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RESULTS OF TEST NO. 3

TABLE C. J

Crack growth testing with low frequency loading only, 60 se¢ hold time, and R ≈ 0.1

Time (sec)	Crack Length (in.)	AK (pei in. 1/2)	Crack Growth Rate (in./sec)
50691	0 3503		
5		22046	è
56931		25661	. 3763E-0
60531	•	6/657	.4100E
15009	9095 D	26775	· .
1070 P		27134	
	0.4051	27862	•
164/0		28819	
1640/		30004	
1031		30259	7010E-0
16612		30661	7309E-0
19757	0.4745	31397	1070E
1604/	•	31502	9375F
16547		31679	1275F
14751	0.4857	32028	1641
75471	0.4986	32776	3170F
75471	0.4985	32745	10112
75951		13700	31041.
76191		ACASE	0.2134E-04
76611		14.340	36667
17091	0.5359		0.21//14
16677		154 67	•
77811			. 2504E
78351		(B70)	2731E
78711		97676	0.2819E-04
78771	•	1905	. 2519E
16197		26192	. 2615E
79856		36809	0.2368E-04
80121		BI JOS	. 2070E
B0421	9719 O	40865	. 2090E
81137		41301	0.2228E-04
81511		42618	0.28545-04
11211		43545	0.3631E-04
	٠	44753	0.4475E-04
76170		46336	0.5109E-04
02432	0.6865	47656	.5146E
9/579		48559	.5207E
1/878	•	50398	
1 5058		51473	5214F
331	0.7316	52877	SSALF
83421	0.7369	53574	
33	765	773	7868E
84081	0.7827	10	10265
			TOTOT -

RESULTS OF TEST NO. 4

Constant uncycled load, high cycle load of various frequencies

Crack Length (in.)	Mean Load (1b)	HF Load (1b)	Freq.	DADN (in./sec)	K Hean (psi ∕in.)	НР К (ра́і <u>√īn</u> .)
0.16	3450	621	110	0	13362.1	3179 7A
0.165	3450	1120	110	7.59600E-07	13583.1	4409.6
0.19	34.00	1450	110	7.85000E-05	14445.6	6160.63
0.213	3420	500	460	0	15476.3	2262.61
0.213	3450	650	097	0	15612	2941.4
0.211	3600	1035	460	0	16205.3	4659.04
0.214	3600	1350	460	5.50000E-05	16333.5	6125.05
C 233	3600	1250	460	1.35000E-04	17136.6	5950.19
	3600	1250	460	1.79000E-04	18424.7	6397.45
	3600	1090	097	1.40000E-04	19741.5	5977.29
	3600	1090	460	1.60000E-04	20978.9	6351.96
	3600	1090	097	1.35000E-04	22231.8	6731.3
0.386	3600	1090	460	1.70000E-04	23512	7118.92
0.415	3600	660	097	8.85000E-06	24786.8	4544.25
0.435	3500	0	760	4.66000E-06	24979.6	0
0.455	3550	007	460	6.13000E-06	26256.8	2958.52
67 0	3550	510	460	1.92000E-05	27944	4014.49
0.522	3650	500	460	1.70000E-05	30424.1	4167.69
0.542	3620	575	097	4.00000E-05	31285.6	4969.39
0.575	3620	575	097	1.02500E-04	33245.1	5280.64
0.616	3550	360	460	1.615005-05	35252.3	3574,88
0.649	3440	200	460	1.83300E-05	36487.6	2121.37
0.671	3405	0	66 0	1.77000E-05	37812.7	0
0.695	3405	•	460	4. 50000E-05	39841.4	0
0.732	3430	0	460	1.17500E-04	43746	0
		F	1 104			
		-	IABLE C.)			
		RESUL	RESULTS OF TEST NO.	ST NO. 6		
		High Cycle Frequency:	Frequen	cy: 460 Hz		
		Low Cycle Hold Time:	told Tim		I	
		LOW LYCLE MAXIMUM K:	uax imum	K: 25 K-141	c	

Ich Max LF K HF K Range (psi /in.) (psi /in.) 27476 0000 6412 000 27637 0000 6412 000 27552 0000 728 0000 25560 0000 7231 0000 26560 0000 7231 0000 26547 0000 7251 0000 26479 0000 7256 0000 24541 0000 7560 0000 24543 0000 7490 0000 24543 0000 7560 0000 24543 0000 7560 0000 24543 0000 7590 0000 24543 0000 7590 0000 24543 0000 7590 0000 24543 0000 7590 0000 24543 0000 7590 00000 245443 00	Growth Rate (in./sec)	0.000099679999	.00012606000	.00014138001	.00015327999	00016482000	018151999	0.00020299001	0.00022772000	0.00022777000	0.00021758000	0000001200000
Hax LF K HF K (Pail XIII) (pai	-			0	0	0	0	0	Ő		0.00	200
	HF K Rang (ps1 /in.	6412.0000	7026.0000	7021.0000	6935.0000	7056.0000	7260.0000	7490.0000	7705.0000	8251.0000		0000 1110
Crack Length (11.) 0.6205 0.6205 0.6259 0.6595 0.6595 0.6595 0.6595 0.6845 0.6845 0.6845 0.7396 0.7396	Max LF K (psi /in.)	27476.0000	26262.0000	26560.0000	26475.0000	25879.0000	24541.0000	24092.0000	23358.0000	23643.0000	21938.0000	10530 0000
	Crack Length (1n.)	0.6205	0.6369	0.6475	0.6594	0.6713	0.6845	0.6966	0.7144	0.7396	0.7560	3 46 E

TABLE C. 5

RESULTS OF TEST NO. 7

		1		
	(10.)		() () () () () () () () () () () () () ((10. sec)
	1 40	2	-	10000
4.7646	0.3496	26180.7734	4619.1836	1000
353.6274	350			10000
	7.5	20094.1445		10000
	7 7		11/2 227	10000
				0000
	7	\$03		00000
	٣.		ó	10000
				00001
				10000
	7.			10000
10		25930.0977		10000
962				10000
902		26044.4453		10000
3		25969.8164		10000
3		25894.2305		10000
F.	-	25941.2773		10000
415.3750	Π.	25890.8359		00001
÷.		25889 . 5586		0000
Ę,		25922.3964		10000
3.3	2.5	1110.6767		10000
		25169.4687	4859.8945	10000
5		24908.0820		10000
5		210		00003
6		25079.3047		00002
2				00002
6:		880		. 00002
-		2		50000
5	•	5		10000
93	•			10000
55		12		
5	•	5		10000
8		3		00002
		279		00000
2		25405 . 6094		.00006
3	3	25569.5586		00001
	3	25305.2305		30000
-	3	25441.1562		. 00005
n, i		23400.4102		80000
5.	3	22242.3308		- 10000
2.5	1	1044.744		
54	14	CTCT 104030		
		2000.0001		
	Į I			
	1.1	2 1		2001074538
-	4	869		
				5C#TTT000

TABLE C.6 (Cont'd)

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													*															
Growth Rate (in./sec)	0.00011676035	0.00011601535	0.00011853676	0.00012063261	0.00012372177	0.00012608935	0.00012877809	0.00012703062	0.00011206591	0.00008711699	0.00005887295	0.00003027947	0.00000974585	0.00000763518	0.00004808255	0.00007431422	0.00010690858-0	0.00011901073	0.00012386656	0.00012535922	0.00012802753	0.00013066530	0.00013706277	0.00015658276	0.00017108464	0.00019060491	0.00021805093	0.00032618269
W K Range (pat /in.)	6143.3750	6104.7031	6221.4023	6261.2969	6345.1562	6391.3633	6604.2812	6660.4961	6607.7109	6810.2422	1521.6460	1309.5159	1396.2622	1480.7656	1703.5078	7168.7266	6686 9102	7220.5000	7265.3516	7204.9531	7307.0391	7313.0977	7659.3750	7854.3320	7552.2773	7714.1992	7799.1172	8066.3203
Han LF R (ped /in.)	24775.5117	24727.0937	24777.2227	24565.5859	24692.2109	24673.5234	24587.4453	24693.5117	24741.9375	24271.8594	24813.1836	25037.1406	24864.1562	24890.7734	27021.7930	27786.2773	1520. 00772	27659.7031	27859.9844	27568.5273	27221.5977	27061.8008	26763.2344	26335.9062	26240.7578	26035.8125	25432.2539	24977.2227
Crack Length (in.)	0.4857	0.4927	0.4997	0.5069	0.5143	0.5219	0.5295	0.5377	0.5463	0.5533	0.5580	0.5602	0.5600	n. 5590	0.5606	0.5643	0.5696	0.5781	0~5862	0.5936	0.6012	0.6084	0.6156	0.6252	0.6348	0.6453	0.6572	0.6669
	24.34.0444	3494.1967	3554.3772	3614.6404	3675.3762	3735.8499		3857.0596	2917.8093	3978.6863	4038.9282	4098.6562	4158.3398	4218.0273	9375.7664	4397.1328	4456.9375	4516.7930	4576.7930	£827. (£84	4697.6523	4758.0078	4818.5273	4479.1953	4940.0430	5001.0977		5123.9727

*Crack growth rate was not based on a sufficient range of crack length.

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TABLE C. 7

RESULTS OF TEST NO. 8

High Cycle Frequency: 248 Hz Low Cycle Hold Time: 10 sec

	FOR CACL	LOW CYCLE Hold Time:	10 sec	
Time (sec)	Crack Lengt ¹ , (in.)	Max LF K (pei /in.)	HF K Range (psi /in.)	Growth Rate (in./sec)
7984.6875	0.2584	21420.8984	787.9695	0.00000941813
9383.9336	4606.0	25230.0312	322.9622	0.00001158264
11084.0156		25599.7070	226.9065	0.00001351392
12386.5664	0.3436	26294.6680	2122.8723	0.00001455599
13163.5195		26434.4062	2349.8621	0.00001497631
14080.9570	0.3707	26770.3984	2298.0520	0.00001488024
14978.3984	•	27066.2187	2610.2822	0.00001415124
15875.2070	0.3963	27487.7969	3066.2178	0.00001307475
16971.7266	0.4095	25899.5234	3429.1816	0.00001222489
18166.2070	0.4219	26296.1211	3541.4712	0.00001361158
19361.3203	0.4363	25894.9062	3646.9678	0.00001984589
20157.6406	0.4558	26672.8047	5140.4961	0.00003097781
20555.8008	0.4675	27050.7266	5286.1211	0.00001640414
20754.8789	0.4750	27109.5391	5368.1836	0.00002467167
20854.1016	0.4515	26361.7422	2574.1467	0.00004010730
23646.2773	0.5050	25651.9141	5539 7539	0 000077/233
23745.5039	0.5127	25658.5820	5607.1758	0.00008925261
23844.0937	0.5218	25790.3672	5824.7070	0.00009586086
23943.3203	0.5312	25939.6523	5838.7500	0.00010083221
24042.5430		25779.1484	5867.9297	0.00010941603
24141.1367	•	25608.0781	6121.9414	0.00011525118
24240.3594	•	25686.8867	6164.8516	0.00012083116
24338.9492		25452.4258	6397.6758	0.00012737671
24438.1758	0.5897	25350.6016	6505.2383	0.00013430661
24536.7656	0.6032	25332.5664	6529.3125	0.00014346938
24635.9922	0.6177	25366.5117	6964.9023	0.00015487349
24735.2148	0.6336	24974.6055	7135.6016	0.00016758432

RESULTS OF TEAT NO 4

TABLE (.8 (Cont.'3)

	Crack Langth (in.)	ан Г. 1 - 1 - 1 - 1 - 1 - 1 - 1 - 1 - 1 - 1 -	EF E Range (pei vin.)	Growth Race (1n., sec.
28517.8006	0.4385	9500.1912	5789.0703	0 00009162258
	0.4500	24008.7461		
28716.4336	9677 0	1176.67665		0.00001536791
9258 · 67950	9584.0	4609.86152	242.213	0 00003636361
34073.0000	0.4876	24674.6523		0.00003904545
34172.1641	0.4913	24764.9023		
			5513.6203	
٠	0.5001		5634.1250	0.00004751662
2352. 69444	0.5049	23527.773		
٠	0.5101	23442.6836	5660.0195	0.00005357758
	0.5156		5917.4409	0 00005561640
· • ·	0.5213	÷.		
· • ·	0.5271	23541.7167	5917.7656	
	0.5330		5897.9102	0.00005977921
35044.4648	1975 0	1982.29262	5918.6133	
35163.5977	0.5450	23860.6094	6027.8047	0.00006349280
35262.8203	C122.0	23610.4687	6012 4883	
33361.4766	0.5582	23440.1289	6088.3516	0.00006944068
35460.8867	0.5453	23640.3672	6212.9570	0.00007149021
	0.5726	23428-2344	6386.4297	0.00007345
	0.5797	23674.5508	6381.5703	0.000076666-9
35756.3066	0.5872	21299.6172	6702.4844	0.00008231173
	0.5953	21322.9687	6806.1328	
35957.0078	0.6046	21509.1016		
36056.1055	0.6146	21455 3945		0.00010989855
36155 2031	0.6257	21362.0696	- 3	
	0.6378	21507.5234		
16351 1750	0.6268	A101 15205	ACTA ATT	0000100100000

AA long duell in the low cycle dominated regime for constant lowfrequency ΔK shows a decreasing crack growth rate.

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(1860) 1197, 6589 11995, 11995, 11995, 1104 22795, 30440 22795, 30440 22795, 2018	CIACK LADGES			
	.9	(per)	a, re	(10./ wc)
	0.2170		379 6965	.0000119302
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	1200 0	•		0.000001100000
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	•	1007 UD624		
		23320.2422	7225 1677	•
15866.2812	•	•		
			•	
17361.2305			2262.0777	•
	0.3218			000004427
				000000
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		23679.1172		21 77L0000
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	1447	•		64
c	•			79/1600000
26726.2656	•	•	4200.9492	.000010000
27025.0156	•	•	1904 . 1171	- 0000122838
	<u>,</u> .	•		
27234 A113	•	•	Ņ	.0000153684
	•		٠	00002236
	•		•	.00003695
	•		2	
	•	328	2	0.00005742149
•	÷.	8	5538.9648	0.00006690415
2217 . 9220 2219 . 9220	•	-	5578.3300	٠.
•	•	2	<u>بن</u> د 1	00001403
/777 - 476	0.4225	23056.6945	5659.2812	0.00007786212

RESULTS OF TEST NO. 10

High Cycle Frequency: 248 Hz Low Cycle Hold Time: 10 sec

Growth Race (in., sec)	00000198970	00000212665	00000264970	.0000322044	1896000000.	.00000333224	22040200000.	.00000307281	.00000274948	00000272366	.00000177671	at tag to out	C000000	.00000271645	.00000322488	.00000301020	.00000261505	.0000245161	.00000281612	000004125	2112220000.	•	86646700000.	.00000767354	.00000661424	.00000615232	.00000556470			٠	.00000/952030	•					.00000766449		. 00000771607			000008800000					÷.	. 30007220265	9766nTnnn.
EF E Range (pet + in.)	1580.7959 0			2331.0115		-	2710.3449 0	2763.6191 0			2920.0320 U 2961.0759 U	0 0110	27.0		242	3509.7227 0	4314					2,52	0742	.6164	.0195	4293.9102 0	3708.1436 0	7747	101		4242.8216 U		274	8680	0312	6097	4570 0	6320	2852 (1916	0781	9099	7010			0547	2404	6414 6602 0 2140 0037 0	1260.6
Han 17 K (pet /10.)	22325.0664		1947.2022		22475.9727		1825.99.3281		•		7420.51022			22766.3047	22859.0352	1626.92162	•		110.85457 10114	23655.9570	23746.0469				23574.0281	23668.5000	23699.8516	23873.2930	23863.7852	23835.9805	2100.04042	24040 1991	23872.3711	23832.6328	23661.8964	23770.6367	23536.5234	23808.5234	23586.0781	23435.2031	23687.6797		23853 . 4648	23951.8359	23955.2695	23726.5391	•	24571.6406	1
Crack Langth (in.)	0.2093	0.2116			0.2269	0.2277		0.27			0.2530	0 3647	•		0.2625			0.279	2/92.0	0.2334				•	0.3146	11 110.0	0.3250		0.3335	0.3390	1031 0	1955.0	9655.0	0.3659	0.3705	•	0.3800		•	0.3960		0.4053		•	÷.	0.4269		0.4497	
	10746.1172		7679.09901	1227.7422	17154.3750	17254.2305	18750.8086				26130.2539	27446 1417	20031.0625	30327.7852					1711.0040	40504.4789	41702.5195					0258.56054			47965.8633	1221		50578.9609	51276.0547	52072.3750	52670.2500	53267.4683	53864.7305	5-561.8242	2115.75655	56053.3437	54650.5859	57147.9647	57744.5781	58241.3281	58738.7109	59335.3203		50011.7552	

TABLE (9 (Cont d)

Growth Rate (in./sec) Growth Rate (in., sec) 81669700001 851595500001 871565120001 55856810001 871562120001 871562120001 HF K Range (pst /in.) 0000000 227.7065 422.0615 422.0615 232.04012 317.1313 320.8130 583.5928 261.5728 261.5728 261.5728 261.5728 261.5728 261.5728 261.6728 261.6738 261.6735 261.0425 5619.0425 5619.0425 5629.0525 5629.0525 5629.0525 5528.7381 6616 4512.230 6612 6612 6612 5528.738 W K Lange (pet · ta.) 6596.3047 6752.2656 7052.8281 7312.5078 7680.3594 7946.0430 605 Hz 10 sec RESULTS OF TEST NO. 11 High Cycle Frequency: Low Cycle Hold Time: Hax LF K (psi /<u>in</u>.) TABLE C.10 26282.022 26282.022 26282.022 26282.0477 26282.0477 26292.26235 26282.26235 26282.26235 Crack Length (in.) Crack Length (19.) 2290.0078 2987.1924 2987.1924 5286.7933 5486.5117 5585.4375 5485.4375 5485.4375 5485.4375 5485.4375 5485.4375 5485.4375 5485.4375 5485.1757 5485.1757 5412.557 54205.1757 6412.2577 54202.1746 6412.2577 54202.1746 54202.1746 54202.2651 54202.2651 54200.5117 64203.5160 64220.2511 64220.25 1 () 1 ()

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TABLE C. II

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RESULTS OF TEST NO. 12

High Cycle Frequency: 605 Hz

TIME	CINCK LENGTH	5	2	GROWTH RATE
(Sec)	(inches)	(psi in)	(psi in.)	(inches/sec)
699.2798	0 5449	41686.8047	443.1199	0.00003621819
599.7678	0 54.89	41366.2500	1004.2280	0.00003664137
699.6240	0.5527	41204.4102	1010.5854	0.00003643995
799.4800	0.5550	41213.6523	1130.5266	0.00003625306
899.3359	0 5593	41035.8164	1440.7424	0.00003724604
999.1919	0 5632	40854.7461	1489.1851	0.00003754467
	0 5671	40848.4180	1536.5725	0.00003780310
	0.5712	40331.3750	1666.4360	0.00003706054
298 7600	0 5747	40315.2617	1677.3757	0.00003474450
398.6160	0.5780	40063.4336	1766.1792	0.0003480908
498.4719	0.5810	39997.0625	2013.2786	0.00003430064
1598.3279	0.5846	39661.4605	1629.3350	0.00003630349
1697.5520	0.5684	39860.0508	2441.6672	0.00003620370
1796.7759	0.5924	39785.5820	2701.9609	0.00003557654
1897.2639	0.5960	39656.7812	2517.5942	0.00003306677
1996.4880	0.5989	39510.9844	2572.6995	0.00003011314
2096.3440	0.6016	1416 64466	2668.1091	0.00002864495
2196.2000	0.6041	39117.0273	2598.4189	0.0002909526
2296.0559	0.6071	39105.0781	2654.8840	0.00003068979
2395.2800	0.6096	38689 . 6602	2667.4263	0.0004080472
2495.1360	0.6150	38890.0586		0.00003817993
2594.3601	0.6193	36960.1797		0.00003600642
2694.2161	0.6201	36927.1016	3488,5422	0.00006604505
2793.4399		36665.2578		0.00008235798
2893.2959	0.6362	36605.7812	3528.1506	0.00008789578
2992.5200	0.6491	35115.4180	3697.1140	0.00008254430
3091.74A]	0.6609	39345.3672	3926.9143	0.00007832758
3191.6001	0.6670	39663.5195	3931.1694	0.00005444091
3291.4561	0.6658	39061.9141	4060.8308	0.00002569969
3390.6799	0.6686	39661.0586	4178.9961	0.00003155028
3490.5361	0.6721	39650.9883	4020.9600	0.00003659808
3589.7600	0.6761	38875.9727	5725.7656	0.00004142890
3688.9841	0.6807	38664.5234	5638.9453	0.00004562846
3786.2C)	0.6854	38612.3750	5647.0698	0.00005017189
3688.0640	0.6900	36171.4414	5803.3164	0.00005480160
396 . 2881	0.6955	37606.6875	5875.2578	0.00006465937
4085.8801	0.7021	37043.4531	6063.6680	0.00007712927
4185.7344	0.7102	36564.9180	6125.2266	0.00009316225
0120 7017				

TABLE C.12 RESULTS OF TEST NO. 13 Constant load, no cycling

K MP K RANGE GROWTH RATE	in) (psi in.) (inches/sec)	.6406 0.0000411751	758 0.0000573215		352 0.0000696139	070 0.00000961069	195 0.00001040402	161 0.00001121570	9219 0.00001191794	125 0.0001290539	758 0.00001347179	320 0.00001386898	0.0001417301	492 0.00001426718	539 0.00001525549	992 0.0001673370	0.0001819010	242 0.0001942298	680 0.00001582147	758 0.00001737023	758 0.00001682866	ō	547 0.00002138640	e	242 0.0002456056	250 0.00002376697	984 0.00002576357	180 0.0002674197	0.0003098156	625 0.00010911345	148 0.00018643984	711 0.00018205498
HAX LF K	(pet t	23961.64	24174.6758	24442.5312	24753.0352	25381.7070	25693.0195	25964.2461	26117.92	26356.3125	26661.6758	27032.3320	27329.1953	27339.4492	27724.2539	27902.6992	25219.9863	28531.8242	28767.6680	26978.1758	29255.1758	29572.7305	30180.0547	30800.1250	31212.8242	31410.1250	32171.3984	32966.9180	32808.2695	34921.5625	40816.2148	39784.8711
CRACK LENGTH	(inches)	0.2869	0.2915	0.2965	0.2996	0.3037	0.3084	0.3127	0.3172	0.3225	0.3285	0.3347	0.3405	0.3462	0.3520	0.3585	0.3660	0.3699	0.3747	0.3770	0.3802	0.3667	0.3986	0.4080	0.4187	0.4274	0.4374	0.4503	0.4476	0.4817	0.5658	0.6039
TINE	(Sec)	11620.3008	12676.3984	13522.0000	13944.3984	14366.8008	14789.1992	15212.3008	15634.6992	16057.1992	16479.6016	16902.1016	17325.1016	17747.6016	18170.0000	18592.5000	19014.8964	19226.1992	19437.3984	19649.3008	19860.5000	20262.8984	20705.3008	21127.6992	21550.1992	21973.3008	22395.6992	22818.1016	23240.6016	23874.8984	24508.3984	24720.0000

RESULTS OF TEST NO. 20

HF K BELLIN 3809.04 3809.04 3809.05 3809.06 38970.07 38970.07 38970.07 38970.05 4040.00 4040.00 4040.00 4040.00 4040.00 40730.05 40530.47 40530.47 40530.47 40530.49 40730.05 4092.49 5015.49 5015.49 5015.49 5015.45 5015.55 5005.55 5005.55 5005.55 5005.55 5005.55 5005.55 5005.55 5005.55 5005.55 5005.55 5005.55 5005.55 5005.55 5005.55 5005.55 5005.55 200 Hz 5 sec ~ RESULTS OF TEST NO. High Cycle Frequency: Low Cycle Hold Time: FF F 131 FF 141 TABLE C. 14 LENGTH FIME 10150.7 117215 117215 117215 17715 17715 17715 17715 17715 17715 224040.1 234040.1 232401.1 2324040.1 232401.1 2 GROWTH RATE INCHESTAC 1.50078E-055 1.55976FC 1.55977E-055 1.55977E-055 1.55977E-055 2.535980FC-055 2.535980FC-055 2.54681F-055 2.54681FC-055 2.54681FC-055 2.54682FC-055 2.546454E-055 2.546454E-055 3.3518746FC-055 3.3518746FC-055 3.3519746FC-055 3.3519746FC-055 3.3519746FC-055 3.452976F-055 5.412097E-055 5.423956F-055 5.423556F-055 5.423956F-055 5.423956F-055 5.42356F-055 5.423576F-055 5.425576F-055 5.425576F-055 5.425576F-055 5.425576F-055 5.4255756F-055 5.4255756F-055 5.4255756F

 F3

 F3
 Hz Sec 8 º LF K 132773.6 132773.6 132773.6 132773.6 135773.7 14177.7 14177.7 14177.7 14177.7 14177.6 13845.6 13845.6 13845.6 13845.6 13845.6 14477.9 14477.9 14477.9 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14478.5 14477.9 14477.9 14477.9 14478.5 14478.5 14477.5 14477.5 14478.5 14478.5 14477.5 14477.5 14477.5 14478.5 14477.5 14477.5 14478.5 14477.5 14477.5 14477.5 14477.5 14477.5 14477.5 14477.5 14477.5 14477.5 144777.5 144775.5 144775.5 144775.5 14478.5 144775.5 144775.5 14478.5 14478.5 144775.5 144775.5 144775.5 14478.5 144775.5 144775.5 144775.5 14478.5 144775.5 144775.5 144775.5 144785.5 144775.5 144775.5 144775.5 144785.5 144775.5 144775.5 144785.5 144785.5 144775.5 144785 High Cycle Frequency: Low Cycle Hold Time: CRACK LENGTH INCLUS 20143 220143 220143 220143 220143 220143 220143 220143 220143 220143 220143 220141 220141 220141 220149 220141 220149 220141 220149 220141 20075 201131 230099 2317999 2317999 2317999 2317999 2317999 2317999 2317999 2317999 2317999 2317999 232444 2327899 232789 24789 24789 24789 24789 24789 24789 24789 24789 24789 24789 24789 2 TIM 171M 171M 171M 171M 172M 19244.7 19441.4 19441.4 19441.4 200413.5 200415.5 200415.5 200415.5 200415.5 200415.5 200415.5 200415.5 20045

GAOMTH FAT INCHES 561 1.6HES 561 3.10515 561 3.10515 561 3.10515 561 3.10515 561 3.12525566 -00 3.32255566 -00 3.32255566 -00 3.33111966 -00 3.3325566 -00 3.3325566 -00 3.3325566 -00 3.3325566 -00 3.323566 -00 3.323566 -00 3.323566 -00 1.4323266 -00 3.323566 -00 1.4323266 -00 3.355566 -00 3.455566 -00 3.455566 -00 3.455566 -00 3.455566 -00 3.455566 -00 3.455566 -00 3.455566 -00 3.555566666 -00 3.555566 -00 3.555566 -00 3.555566 -0

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RESULTS OF TACT NO. 23

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 2 sec RESULTS OF TEST NO. 24 TABLE C. 16

High Cycle Prequency: 200 Hz Lov Sycle Hoid Time. 5 sec

		2	CONUTU DATE			į į	[[
CRACK LENGIN	ן ניייני ניייני			RECONDS	INCHES	21-15d	NJ-184	INCHES/SEC
			1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	4606.84	.223925	15594.7	3179.65	5.056285-06
	1.0/061		2.11230E-00	5409.16	.229521	15641.2	3244.61	5.207936-06
124467	/****	3442.74		6411.5	.234768	15729.5	3296.21	5.201356-04
.261231	C . E/84 I	05.0055		7414.48	.239853	15763	3413.37	4.890755-04
. 2659	14783.9	3666.64	4.04708E-00	8814.3	.245484	15057.4	3482.07	
.272048	14945.7	3720.24	5.23221E-06	11021.5	.253005	14845.3	1400 0D	1 14444F-04
.277987	14825.9	3745.23	5.989456-06	13029.8	258689	14948.7		00-Ur00r0.F
.282363	14711.2	3756.07	6.09794E-06	14834.9				2 101101 - 00
.287946	14573.8	3763.32	6.67876E-06				24.0047	3.643346-06
.294741	14706.1	3621.41	6.745eBE-06			1441	10.101	4.24272E-06
. 3009 49	14499.9	3858.98	6.30215E-06		204/2.	14676.8	3868.84	4.72856E-06
TOPOL -	1 2 7 0 4	1015.08	5.858375-06	18049.4	.280763	14458.4	3850.18	5.27330E-06
			5.41400E-04	19651.8	.286362	14493.5	1939.07	5,595616-06
0/1012.	0.04041	10.1/10 10.1/10		20654.4	.292063	14514.9	3946.46	5.94446-04
.314487	14668	18.C246	4.2078/E-00	21457.1	207023			
.332961	15232.4	4189	4.62325E-06				2000 C	6.21YOBE-06
SESEE.	15149.5	4297.49	9.28250E-06	+ 00+77	414707.	14338.8	3982.93	6.53443E-06
110517	4.1794		1. 1195AF-05	23263.1	.308732	14465.2	4032.49	6.86742E-06
				24265.7	.315691	14307.5	4077.22	7.332856-06
700717.				25068.4	.321781	14406.5	4088.78	7.704115-00
	0.73811		10 UCC/ 10 - 1	25869.9	.327979	14024.1	4166.57	B 1 75.016-04
F09409.	7* 48/47	10.4704		26672.2	.37482	14311.8	4170	
	14828	C/ · 2104		27474.8	7541937	E.11141	42.44.74	
100007 ·	14966.2	4570.06	2.040055-03	28077.6	347577	1 2 1 7 4		
.370154	14534	4724.36	2.24713E-05	28480 4				
.375874	14808.5	4853.29	2.541946-05			4 · A0741	00.0724	1.138736-05
.381114	14747.4	5033.01	2.5J292E -05	1.10400		0.00141	4481.99	1.203026-05
.386873	14460.6	5085.29	3.08084E-05	0.004.17	A04407 .	14101.1	4576.48	1.289886-05
. 191744	14422.7	5145.43	3.30694F	29884.4	. 369844	13986.7	4556.17	1.41380E-05
100500	I AAIO I			30285.5	.37476J	14056.1	4240.04	1.615856-05
E SOLOT		100 VE		30887.1	- 384484	15062.1	4966.88	2.44796E-05
10100V				31288.2	.394678	14908.8	5371.99	3.71809E-0 €
450414	0 717F7			31488.5	.401815	14949.7	5480.04	A.52557E-05
				31688.9	.411762	15071.2	5628.37	5.086895-05
				31887.3	.423788	14960.5	5782.32	5.44571F-05
11011	1 1040			32089.8	.435124	15170	5918.37	1000 10 10 10 10 10 10 10 10 10 10 10 10
				32290.1	- 447317	15193.8	6058.01	A. 337A76-05
	5 · 01 4 • 1			32490.8	.46023	15178.9	A250.05	
.446066	1.17021	2816.3	2.3014/E-05	32692	474052	15174.2		
. 453645	15142.6	5840.14	5.4666E-03	12807				CO- 30707 · /
.459035	14841	5846.14	5.453°8E-05			0.0001	6021.06	8.15655E-05
.464295	14727.1	5883.31	5.803396-05	0 10000	404000.	15222.5	6880.98	8.83606E-05
•				1 · 7 A 7 7 7	- 224824	15302.8	7195.33	9.548495-05
				9.4445	.544718	15342.5	7489.8	1.05131E-04

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RESULTS OF TEST NO. 26

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 10 sec

GROWTH RATE IMCHER/GEC	2.40107E-05	2.43103E-05	2.321466-05	2.25147E-05	2.27702E-05	2.29000E-05	2.21918E-05	2.22762E-05	2.091596-05	1.94536E-05	1.75716E-05	1.56418E-05	1.39363E-05	1.218656-05	1.07715E-05	9.79167E-06	9.58591E-06	9.61611E-06	6.58685E-06	4.52905E-06	6.73840E-06	1.18570E-05	2.27630E-05	3.46956E-05	4.091396-05	4.38715E-05	4.96152E-05	5.34657E-05	6.34540E-05	6.74531E-05	7.172186-05	7. 87511E-05	8,33411E-05	9.00567E-05	1,00405E-04	1.10882E-04	1.21563E-04	1.325356-04	1.42347E-04	1.56928E-04	, 266E-04
	348.538	370.757	372.199	400.22	416.665	427.059	444.469	435.19	1025.44	1045,55	1054.79	1078.61	1094.12	1103.8	1120.6	1127.71	1167.49	1192.79	1180.33	1828.52	2052.56	22.2404	4287.45	4375.27	4487.82	4530.77	4660.48	4706.52	4831.52	4919.57	4990.77	5085.51	5152	5263.77	5364.22	5494.63	5614.01	5767.28	5937.38	6137	6339.12
	20446.3	20730.1	20465.8	20749.1	20766.5	20909.5	20868.7	20951.2	21051	21135.3	21263	21337	21538.5	21475.1	21536.2	21685.7	21350.2	21726	21350.7	20889.9	20592.2	20497.5	20474.3	20348	19998.9	20268.7	20406.3	20460.6	20498.3	20644.6	20570.2	20661.2	20651.6	20474.1	20701.8	20440.6	20543.7	20857.3	20941.3	21080.6	21326.4
CRACK LENGTH	245289	.272813	.279544	.286124	.292666	.29948	64E90E.	EEOTIE.	.317605	.323834	.32927	.335894	.341657	.34661	1352082	.357041	.364605	.37402	.386309	.414256	.419472	.429797	.43503	.438857	.446038	.45067	.460062	.464475	.475906	.482789	.48982	.497457	. 504959	.513324	.522867	.533467	.545155	.557852	.571526	.586322	.603259
TINE Beconda	1387.89	1664.83	1981.78	2279.08	2575.89	2872.87	3170.42	3368.47	3669.29	3970.15	4271.24	4672.2	5073.03	5474.5	5975.8	6578.16	7280.9	8083.49	9889.53	21026.1	22129.9	23334	23735.3	23936	24137.2	24237.6	24438.4	24538.8	24739.5	24839.9	24940.1	25040.5	25141.4	25241.7	25342.2	25442.5	25542.8	25643.2	25743.6	25844	25944.3
							GROWTH RATE	INCHES/SEC	9.66318E-06	1.19874E-05	1.44594E-05	1.79219E-05	2.20731E-05	2.72575E-05	3.33616E-05	3.47997E-05	3.66119E-05	3.82572E-05	3.94627E-05	4.14606E-05	4.38213E-05	4.62507E-05	4.84076E-05	5.04204E-05	5.26406E-05	5.52307E-05	5.83182E-05	6.31815E-05	7.12344E-05	a.51581E-05	1.03244E-04										
	MU 25	. 1	y: 200 Hz				¥ ۲		4183.33	4336.9	E.9744	4703.94	4754.94	4824	4959.78	5040.66		5149.12	5224.06	5293.93	5364.33	5436.1	5499.68	5612.34	5716.72	5820.48	5891.86	5977.56	6406.62	6742.58	7157.55										
TABLE C. 17	PRSINTS OF TEST		High Cycle Frequency:	Low Cycle Bold Time:			ا ج	PSI IN	15232.4	14956.2	15104	15076	14992	14754.2	14656.9	14789.1	14923	14611.7	14588.6	14841.9	14564.1	14551.3	14290.3	14338.3	14254	24142	13969.1	14082.9	14282.7	13962.8	14004.1										
			214	9			CRACK LENGTH	INCHES	59591910.	.32126	.327102	.JJ1765	545455 ·	.344727	.354068	.359804	.366015	.371875	.377934	.384462	.391241	.398544	. 406035	6E6E14 ·	.422356	. 430881	.4397	.449067	.458976	.470238	.484288										
							TIME	SECONDS	3610.3	4251.17	4731.53	2022	5372.7	5492.85	6013.3	6173.6	6333 . 65		6 6 6 5 3 . 9	-	6974.95	7135.05	7295.3	7455.6	7415.85	7776.05	7936.25	8094.35	8256.35	8414.6	6276.85										

З, RESULTS C' LEST NO.

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82 1 cycle Frigu roy: Cycle Hold Time: High .

RESULTS OF TEST NO. High Cycle Frequency: Low Cycle Hold Time La constant a constant HLENGTH CRACK LE 20046 231334 231334 231334 234384 234384 234384 234384 234384 237385 23738 117.4% 117.5% 73 74 Η, LENGTH CRACK (F CRACK (F CRACK (F 279175 279175 279175 279175 279175 279175 279121 279121 279121 279121 279121 279125 291222 2012229 201229 2012229 2012229 2012229 2012229 2012229 20122 TIM 9000.0 9000.0 9100.0 9

GROWTH RATE 1.754555 1.754555 8.459456 8.459456 8.459456 8.459456 8.504526 9.965557 9.965557 9.965557 9.965557 9.955556 1.117426 1.117426 1.117426 1.117426 1.117426 1.117426 1.117426 1.117556 1.125568 1.1175568 1.125568 1.125568 1.125568 1.125568 1.125568 1.125568 1.125568 1.125568 1.125568 1.125568 1.125568 1.125568 1.155588 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.558558 1.55585858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.55585858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.555858 1.55585858 1.55585858 1.55585858 1.555858 1.555858 1.5558

TABLE C. 20

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RESULTS OF TEST NO. 31

RESULTS OF TEST NO. 30 TABLE C.21

High Cycle Frequency: 200 Hz Low Cycle Kold Time: 5 sec

	Migh Lov C	Migh Cycle Frequency: Low Cycle Wold Time:	200 Hz 5 aec			¥ 3	High Cycle Frequency: Low Cycle Noid Time:	ncy: 200 Hz me: 5 sec	
TIME	UTATU I EVOTU	1 1 1			11ME Ger	CRACK LENGTH		HF K	GROWTH RATE
SEC	INCHES			GROWTH RATE			MT 4 194	N17 184	INCHES/SEC
2534.11	.270249	21384.1	ICC OFF		2674.39	197455.	29768	400.440 510.745	1.57077E-05
1221.94	.203215	21409.7		1.524316-05	3221.53	.355067	30096.6	504.885	1.47550E-05
5125.49	.304068	21891.2	177.014	1.40442E-05	3776.39	.34273	10440	1327.95	1.37997E-05
6022.22	. 31441	22158.4	207 - 707	1. 10034/96-10	4485.42	.372168	30372.2	1365.15	1.25640E-05
4919.44	.320192	22111.4			5295.14	.381229	30149	1395.49	1.090536-05
7664.46	166802.	20221.7		1.241736-00	4105.77	.389321	30943	2239.28	9.76339E-06
905 8 .19	. 354705	6.4261	390.154		7018.19	89795.	30826.2	2327.9	8.9329BE-06
10103.3	.344402	20024.9			8133.33	.4097 85	30716.4	2366.76	B.22917E-06
4.4001	.374934	4.7941	415.115		9240.41	.414905	31343.1	2409.42	8.33220E-06
	552485.	2.979.5	417.489	7.83000/E-00	10262.4	.422804	23008	2668.94	9.395636-06
13262	.193241	19809.1	1522.25	1 1 1 2 0 4 E - 0 0	9.72211	.43293	30334	3444.71	1.16632E-05
14717.9	. 406799	20180.4	1571.52		11665.3	.439168	30034.3	4098.53	1.37667E-05
20327.1	.419784	20390.1	1451.33		12545.5	N48844 .	29869.7	4267.44	1.61338E-05
24803.5	. 43298	20437.8	1951.45		13052.7	.437479	30083.3	4401.18	1.84552E-05
2003.6	141744.	21010.3	2473.5		13459	. 465075	29684.7	4451.28	2.13422E-05
31052.8	100404.	21001.6	2741.44	2 - 1 - 1 - 1 - 1 - 0 - 0 - 0 - 0 - 0 - 0	13916.6	.475411	29820.1	4559.26	2.52906E-05
34254.2	. 447068	21042.4	2780.87		14220.8	.483282	29812.9	4658.13	2.79442E-05
37912.2	.463746	20146	752.401		14525.6	.492122	29922.5	4714.49	3.06652E-05
D . 80747	. 491487	19936.3	517.393	1.7/800E-08	14779.8	. 500603	29624.6	4750.7	3.353016-05
41252.8	-201109	19470.5	535.271	0-102 102 -00	15034	.509234	29303.2	4806.47	3.56431E-05
42749.3	.51004	E.90641	352.054		15288.1	.518225	29083.2	4914.85	3.786556-05
	.522503	19747.7	545,445	0.90175-00	15541.6	.128346	29155.3	4965.07	4.04528E-05
4.61/64	199900.	20310.7	4412.4	1 . 71.231F_06	15795.8	.539072	29219.7	5024.22	4.412156-05
46070.1	.54262	D. E4991	4433.72	2. FEADTE - 03	15999.2	.548211	29334.2	5118.24	4.70995E-05
0.4/70F	467640.		TE. 1944		16202	.557866	29039.4	5229.38	5.02227E-05
10028-2	-141655.	_	4774.45	10-14/10/00 10-14000007 - M	16405.5	.568476	28879.1	5341.94	5.37809E-05
6 · 78894	840492.		4874.7	10-308/2/	14408.4	.579901	28620.4	5427.09	5.64278E-05
4 . CBU / 4	.57746	1	4848.79		16761	.588624	28378.3	5476.01	6.05860E-05
47288.2	.586715			10 11 11 11 1	16913.1	.597864	28478.2	5582.71	6.52224E-05
47491	.596852	20058.9			17065.9	.407898	27996.2	5658.76	7.17634E-05
8.54474	. 604851			3,1027UE-05	17167.6	. 61383	28468.4	6020,88	9.21304E-05
2.99714	262219 .			00-348478-7	17319.9	. 630383	29053.5	6572.11	1.00003E-04
47948.4	.422978		5407.44	0.1734/E-05	17421.9	.641767	28627.6	6840.15	1.09941E-04
48100.7	.634079		5440.25	0.82441E-05	17523.5	.653591	28513.9	7173.98	1.17643E-04
		•	74.200	50-377/77·/	17726.6	.679745	27846.9	7694.4	1.25042E-04

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RESULTS OF TEST NO. 32

High Cycle Frequency: 200 Hz Low Cycle Nold Time: 5 sec

TABLE C.24

200 Hz 5 mec RESULTS OF TEST NO. 33 Bigh Cycle Frequency: Low Cycle Bold Time: CRACK LENGTH INCHES 5513505 551152 551152 551152 551192 551192 551192 551192 5513256 5513256 5513256 5513256 613326 613326 65132 653864 65712 65712 65712 65712 65712 65712 65712 65712 65712 65712 65712 65712 65712 704119 891.597 1190.28 11994.47 1898.47 1898.47 1898.47 33252.78 33252.78 33252.78 33252.78 33252.78 33252.78 33252.55 4127.5 TIME BEC OROWTH RATE INCHES/SEC HF X PSI XI 1771.11 1771.11 1989 1771.11 1989 22204.12 22344.12 22344.12 22344.12 22374.12 22374.12 22374.12 22374.12 22376.12 22376.12 22376.12 22376.12 22376.12 22376.12 22376.12 22376.12 2328.03 2328.04 2328.04 2358.02 240.02 240.02 240.02 240.02 240.02 240.02 240.02 240.02 240.02 240.02 25 LF K PSI /IN 30228.5 30228.5 30228.5 30228.5 30735.7 30491.1 30491.1 30491.5 30577.5 30777.5 30777.5 30777.5 30777.5 30777.5 30777.5 30777.5 30777.5 3 LENGTH CRACK LE IMCHES 3791748 3791748 3791748 404456 414456 414489 444489 444489 444489 444489 444489 444489 5524253 5524253 5528165 5528266 5538846 5538846 5538846 5538846 5538846 5538846 5538846 5538846 5538846 5538846 5538846 5538846 5538846 5538846 5538846 5538866 5538846 553886 5538866 5538666 5538666 5538666 5538666 5538666 5538666 11 ME 2957 2574.9 2524.9 2524.9 2524.9 25291.9 7571.9 7571.9 7571.9 7571.9 7571.9 7571.9 12323.1 1136366 113636 113636 1136366656 1136366666 11366666666 1136666666

2,94116E-05 2,499391E-05 2,499391E-05 2,49938E-05 2,49038E-05 2,41725E-05 3,12651E-05 3,12651E-05 3,12651E-05 3,12651E-05 4,11327E-05 5,18052E-05 5,18052E-05 5,18052E-05 5,18052E-05 6,86088E-05 1,00388E-05 1,441102E-04 1,15192E-04 1,53236E-04

HF K P81 VIN 14672.72 14672.72 1472.72 1472.72 14842.43 1884.43 1884.43 1884.43 3065.15 3065.15 3065.45 3065.55 3065.45 3065.55 3065.45 3065.55 3065.45 3065.55 305.55 3

BROWTH RATE INCHES/SEC

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RESULTS OF TEST NO. 34

Migh Cycle Frequency: 200 Hz Low Cycle Hold Time: 5 sec

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 5 sec

RESULTS OF TEST NO. 35

TABLE C. 26

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CRACK LENGTH	INCHES		.220252	.229704	.238834	248212		776/07.	.266295	.277634	.287598	. 296608	.307081	.316761	.326537	.336814	5146413	.355346	.366256							
																			16097							
																		-								
GROWTH RATE	INCHES/SEC	2.23904E-05	2.20114E-05	2.20334E-05	TOTAL C		CO-3705/4.1	1.873036-05	1.79203E05	1.74827E-05	1.73420E-05	1.479436-05	1.45548E-05	1.450456-05	1.494495-05	1.803375-05	2. 14741E-05	1. 357375_A5	A. AATO15-05	445295-05	B.30214E-05	1.09585E-04	1.27637E-04	1.474336-04	1.70636E-04	
		572.952	570.943	587.689				2378.97	2432.46	2730.55	2754.37	2892.72	3045.8	1095.11	3173.42	52 . VAET	TAR7.75			5145.45	4008.17	6429	4810.44	7009.25	7465.49	
رد الا دور الارتيار	NT A TOL	39264.7	9.20001	40044			0.444.0	39520.7	39414.9	39209.9	38734.7	38627.3	38310.2	37576.5	37811	37720.5	17977.7	Tecer	38897.1	9.573.9	38967.3	38116.4	37972.4	37984.6	36834.4	
CRACK LENGTH		.520776	.527852	534848			244100.	.559348	.566893	.574184	.581214	589244	618040	402398	61179	417424	8012CY	LAOFTA.		.448211	65456	• 663386	.669264	.474018	. 683478	
TIME																										

GROWTH RATE INCHES/SEC 1.48558E-05 1.48558E-05 1.48558E-05 1.43793E-05 1.43793E-05 1.43732E-05 1.34332E-05 1.23432E-05 8.72545E-06 8.72545E-06 8.72545E-06 8.30781E-06 8.30781E-06 8.30781E-06 8.30781E-06

HF K PSI VIN 313.35 313.35 324.477 334.477 335.257 335.257 335.257 335.257 1120.657 1120.64 1120.64 1120.64 1120.64 1120.64 1120.64 1120.64 1120.64 1567.66 1968.76 1967.86 1998.74 1998.74 1998.74

TABLE C.27

RESULTS OF TEST NO. 36

Nigh Cycle Frequency: 200 Hz Low Cycle Mold Time: 2 sec

GROWTH RATE INCHES/SEC	6.24181E-05	5.08084E-05 4.54797F-05	3.749736-05	J. 75249E-05	3.59084E-05	3.46251E-05	5.85767E-05	9.51374E-05	1.30634E 74
HF K	794.299	665.92 646.157	614.271	3818.71	4023.53	4199.52	4258.6	4559.42	5064.76
LF K	36239.4	40318.8 40687.7	40479.3	41010.8	40642.2	40870.7	39471.7	41145.1	43895.2
CRACK LENGTH INCHES	.52279	.545328	.560293	.576002	.563788	.598895	.599264	.629442	.676014
TIME SEC	582.251	975.532 1173.72	1373.04	1787.29	2000.23	2429.35	2863.25	3301.49	3743.34

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				GROWTH RATE	INCHES, SEC	1.34865E-05	1.517386-05	1.49006E-05	1.41691E-05	1.33242E-00	1.200626-05	1,13862E-05	1.07736E-05	1,003906-05	9.0/3/36-00 0.000875-06	B.07910E-06	7.88914E-06	6.26715E-06												GROWTH RATE INCHES/SEC	5.60387E-06	5.81127E-06	5.32295E-06 4.97156E-06	4.46742E-06	2.37025E-06 05407E-06	3.14442E-06	5.89327E-06	1.325906-05	5 / 5100E - 00					
		39	200 Hz 180 sec	ц Ц	PSI VIN	307.762	299.098 788 788	307.132	312.66	331.125	402.849	417.871	1143.38	1113.61	1128.57	C0.4/11	18.1121	2595.84							. 40	-11 000	200 Hz	Jes noi		HF K	1988.55	2069.45	2217.8 2232.79	3247.91	3116.94	4356.08	5008.89	5310.19	5200.J	84.05/C				
TABLE C. 30		RESULTS OF TEST NO.	High Cycle Freugency: Low Cycle Hold Time:		PSI VIN	20318.6	20190.5	20143.5	20025	20024.3	19962.3	19794.7	19917.2	19930.5	19964.1	19941.2	1.4441	1.5572.1					TABLE C 31	TO THOM	RESULTS OF TEST NO.		High Cycle Frequency:	LOW CYCLE HOLD IIme:		TH LF K PST JN	20803	20791.8	20952.9 20856.5	19397.9	19536.1	19913.7	199 10.7	19820	20564.6	+·C/107				
		121	H1gh Lov (CRACK LENGIN	14222.	.235284	.247346	.263852	.273637	.282697	-27122.	-306193	.317269	.326865	.337612	426242.	570C71									H1gr	PON		CRACK LENGTH	.224916	.232454	.244775	.350457	.368165	.37881	.405611	.415805	423876	, 428343				
					TINE	3061	3601	4321	4861 5401	6121	6841	7562	2828	2006	10984	12066	12967	13688	01001											TIME	4862.16	6124	8466.53 10628.8	27383.6	33149.8	4 • 95460 • • 85400	48466.4	49367.7	49728.1	49908.2				
				GROWTH RATE INCHES/SEC	1.282456-05	1.22103E-05	9.77332E-06	8.72546E-06	8.21591E-06	8.05243E-08 8.15975F-04	8.95050E-06	1.04905E-05	1.23492E-05	1.40288E-05	1.741815-005	1.92647E-05	2.07186E-05	2.30221E-05	2.60315E-05	2.96901E-05 3.4/8/85-05	4.05215E-05	4.33337E-05	4.7041BE-05	5.091586-05	5.37433E-03 5.04511E-03	6.45450E-05	7.071196-05	7.93374E-05	8.72410E-03 1.01465E-04								INCHES/SEC			4.246266-05	4.15735E-05	3.96830E-05 7.030365-05	4.13657E-05	
	-	(M) H.	•	HE K PSI JN	285.125	308.389	26.010	1308.29	1003.53	20. 2011	1364.53	1396.94	3505.26	3714.31	74.2CV2	4210.14	4321.05	4435.76	4285.71	4211.27	62.974	5154.63	5297.31	5388.45	5530.66	5957.54	6148.06	6289.2	6684.8 7193.5				. 38	: 200 Hz	2 sec	ļ			10.01	784.078	2261.8	2377.18	5750.75 6246.45	
	RESULTS OF THIT M	High Frank	town of such a such as the	LF K JIN	20412.8	20406.8	1.0000	19948.2		20392	20517.5	20686.8	20782.2	20778.3	20792.9	21014.8	20778.5	21047.5	20623.3	20727.8	21453	21319.1	21356.7	21255.3	21601.1	21342.9	21051	21590.2	21367.3 21432.5		TABLE C.29	4	RESULTS OF TEST NO	High Cycle Frequency:	Low Cycle Hold Time:				54156.4 18414	37952.7	37148.4	37049.2	35904.7	
	œ	40.4		CRACK ! ENGTH INCHES	.201.42	.209973	0C0417.	.244773	.254139	.263076	.280498	.290896	.300853	.309817	.320294	. 340845	.350345	.360387	.370187	.381638	014445. 000014	619916.	42887	.43908	.450168	475941	40000	504619	.522147					High	Low		CRACK LENGTH		10100.	26E129.	.681145	. 690337	100440.	
				TIME SEC	2406.93	3017.27		6316.86	7442.93	8576.71	7/10.42	12016.7	12946.3	13649.2	14357.5	-0/061	2.150A1	16510.2	16936.1	17361.9	1.787.1	18424.5	18639.1	19852 . 6	19045.3	19491.8	C 70201	19918.4	20130.7								¥11		8001/30 1010 83	1289.97	1523.46	1760.19	2492.92	,

			GROWTH RATE INCHES/SEC	1.551466-05	1.47507E-05	2.080546-00	2.98083E-05	3.40959E-05	3.77073E-05		4.87286E-05	5.032536-05	5.11316E-05							LICS HIHOS	INCHES/SEC	8.55542E-06	1.35011E-05	1.55682E-05	1.55457F-05	1.399106-05	1.25462E-03	1.48081E-05	2.11307E-05	1.86422E-05	4.000366-03	3.351536-07 7.184095-05	20-34001.V	
NO. 43	500	: 180 sec	HF K	4426.15	4637.38	00.0C/F	66.4E6¥	5088.47	5265.25	E7.0042	6098.45	6322.17	6485.63				40. 44	7: 200 Hz		× 11	PSI IN	٠.	2606.96	2869.61	78./705	3362.06	3421.96	3630.83	4112.72	44.5Y . /2	74.04R4	01-98-10 5254.81	5351.43)
RESULTS OF TEST NO.	High Cycle Frequency:	LOW CYCLE Hold Time:	H LF K	16069.6	15291.4	10100.7	15640.6	16278.9	15890.2	14682.7	16372.6	16533.5	16784.9			TABLE C.35	RESULTS OF TEST NO.	High Cycle Frequency: Low Cycle Hold Time:		1 K		41012.3	41259.7	40159.8	40144.1	40071.3	39968.2	2.11E6E	40630.4	1.44004		42630.2	40749.3	
	Гн 1	07	CRACK LENGTH INCHES	.270508	.281753	12662	309112	.320845	790925. 7485442.	.363509	.397238	.415276	.424225					H1 Lou		CRACK LENGTH	INCHES	-513395	.523347	.529613	.547145	.557079	.565103	.577035	140/9C.	040747		.662041	.677036	
			I TIME SEC	6276.52	7352.14	8428.43	8786.94	9145.48	9863.11	10221.7	10938.8	1129B	11477.3							TIME	SEC	7165.76	8241.41	47.778	9676.08	10393	11110.6	12186.2	13620.4	14131.1	16310.4	16489.6	16668.8	
			TTAG UTHOGO	INCHES/SEC	1.69518E-05	1.00010E-05	2.06432E-05	1.85967E-05	1.28805E-05 7 864766-05		1.68599E-05	2.54616E-05	3.65447E-05	5.09255E-05 4.41874E-05	7.89584E-05							GROWTH RATE	INCHES/SEC	5.35195E-06	0.87047E-00		2.32200E-05	2.67585E-05	2.99493E-05	3.23090E-05	00-0000tt.0	3.645/1E-05 7.83975F-05	A. 074485-05	4.33959E-05
	41	200 Hz 180 sec	y J	NIA 154	510.935	529.656	2149.6	2500.81	2867.2	2915.88	3772.74	4457,26	4832.91	5614.28	5891.85			42	200 Hz	180 sec			NTA TRA	4321.8	14742 1470	83.044	4763.67	4831.52	4900.5	5080.98		64.102C		5663.06
TABLE. C. 32	RESULTS OF TEST NO.	High Cycle Frequency: Low Cycle Hold Time:		NIV 154	20400 20953. 5	31156.8	30271.8	30372.3 TATA	30354.1	29900.3	20379.3	30861.4	29704.7	30033.7	29872.8		TABLE C.33	RESULTS OF TEST NO.	High Cycle Frequency:	Low Cycle Nold Time:			NTA 101	14928	/ 1/001	15968.9	15923.2	15739.4	16227.6	16205.7	1 20401	16680.5	14329	14708.3
	21	H1gh Low (CRACK LENGTH	INCHES	.472873	.485923	-506368	00142C.	.5438	- 223017	.548245	A47//C.	40094 ·	.609319	.620827			24	High	Lot		CRACK LENGTH	C SHOW	.346051	100400.	511646.	.384119	.39279	.403837	971214	41041	655554 ·	467475	. 48248
			117E	5EC 12069.1	12969.1	13509.2	15131.4	14031.9	17612	16896 76100 0	20731.1	21098.5	21465.5	21649	21832.4	146						TIME	365	36470.J	40047.8	60787.1	61147	61506.7	61866.2	A . C7770	07045.0	6-300°.6	1.29454	44024.8

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ARSULTS OF TEST NO. 46

High Cycle Frequency: 200 Hz Low Cycle Hold Time: 2 sec

RESULTS OF TEST NO. 47

TABLE C. 37

High Cycle Frequency: 200 Hz

3	10 sec
ineur y.	Low Cycle Hold Time:
	Hold
	Cycle
	3

GROWTH RATE	LINCHE SA SEC A - SAGTAF - JA	3.65665E-00	3.37474E-06	3.23888E-06	3.19975E-06	3.25549E-06	J.64731E-00	4.66809E-06	7.72139E-06	1.26483E-05	1.96343E-05	2.56312E-05	2.95346E-05	3.35075E-05	3,537486-05	4.30391E-05	4,20667E-05	4.50501E-05	4.87157E-05	5.23991E-05	5.72547E-05	6.29001E-05	7,042916-05	7.84051E-05	8.74923E-05	9.20887E-05	9.73514E-05	1.03350E-04	1.08453E-04	1.13526E-04
HF K BC L L	151 151 190.45	1151.54	1191.84	1184.59	1608.21	1596.01	2384.34	2461.62	2636.87	3008.95	4083.25	4161.16	4358.64	4422.33	4519.87	4669.44	4725.68	4819.65	4894.85	5029.51	5074.79	5316.53	5524.83	5636.9	5880.48	6026.09	6204.64	6315.1	6482.96	6596.82
LF K Ber Th	0.05.00	29645.8	29927.9	29790.2	29804.5	29969.8	29730.4	29663.7	30160	29584.8	29254.6	26971.4	29231.8	29918.4	29014.1	29330.3	29208.1	29701.1	29954.6	28504.7	28770.8	29543.9	28954.1	29985.7	28480.3	28966	28805.6	29297.3	28853.7	28520.9
CRACK LENGTH	.331421	.349872	.357237	.36623	.37468	.383359	.390238	25046E.	.412084	.41973	.427309	.435027	.443211	. 453052	.460127	.471618	.480364	.469113	.498612	.508798	.519764	.531742	.545126	.560165	.576952	.586211	.595864	. 606048	.616676	. 628268
TIME	35651.17 5461.17	10114.9	12366.4	15232.3	18099.3	20863.6	23014.1	25370	27827.8	28749.8	29364.3	29774.4	30081.7	30388.9	30594.3	30901.5	31106.3	31311.2	31516.4	31721.3	31926	32130.9	32335.8	32541.1	32746.1	32848.6	32951	33053.5	33155.9	33258.3
	GROWTH RATE	7 718845-04	7.544705-04	7.22277F-06	5.42116E-06	A.38648E-06	6.42596E-06	7.169886-06	8.14466E-06	9.36793E-06	1.15481E-05	1.375876-05	1.57161E-05	1.774385-05	1.9797AF-05	2.11949F-05	2.29015F-05	2.49247E-05			2.15794F-05	3.41831E-05	3.474525-05	10-10-00-01 M	A.018075-05	4.2141F-05	4.79207F-05			
L L		770 707		1840.47	1987.6	2164.03	2421.05	2685.74	3366.67	3607.5	3675.86	3840.79	4155.86	4280.29	ATAA. AR	4509.91	4580.21	AX 17.94	4801.47	4944.51	2021.03	5151.52	CA. 71E2	5415.38	5502.31		101.047	- /11.75		
2		TOARU E		1.95401	30209.2	30264.7	30124.3	29734.5	29567.9	29523.5	29443.3	29295	7.EEE62	20249.5	20235.4	20274.1	20211.1	20077.4	10040	20204.5	2880A	28673.4	28551.5	28544.9	28554.5	24514.2	28401.2	20407.0		
COACK ENGTU	INCHES	11011			10700	00000	155114	52375	432534	439094	449634	80404 ·	446197	ATARIA	45004	191994						A732A	20095.	SA98AA	409149	404729	415440			
		30 1210	C7.1084		7.80211	9.19121	1.1031.B	17916.3	19172	20010.8	21058.4	21897.4	22526.1	27155.1	7.79710	24201.5	C. FCAAC		1441 B	25881.4	101.AC	24720.7	27140.5	27350.3	27540.1	27770	77979 B	20100.4		

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