DEFLAGRATION TO DETONATION TRANSITION BEHAVIOR OF ALUMINIZED HMX

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**Title:** Deflagration to Detonation Transition Behavior of Aluminized HMX

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**Abstract:**

The effect of fine (5μ) and coarse (95μ) Al on the deflagration-to-detonation (DDT) behavior of HMX, in charges pressed to 90% theoretical maximum density (TMD), was studied. It was found that Al slows down the DDT process increasingly as the content of Al increases; that is, it increases predetonation column length and the relative times to detonation. Fine Al is more effective than coarse in delaying DDT, at a given content, but both are inferior to wax in this respect.
SUMMARY

The effect of fine (5μ) and coarse (95μ) Al on the deflagration-to-detonation (DDT) behavior of HMX, in charges pressed to 90% theoretical maximum density (TMD), was studied. It was found that Al slows down the DDT process increasingly as the content of Al increases; that is, it increases predetonation column length and the relative times to detonation. Fine Al is more effective than coarse in delaying DDT, at a given content, but both are inferior to wax in this respect.

This report was prepared under Task ZR0130901, IR-159. The present results and conclusions on the transition from burning to detonation of aluminized HMX explosives should be of interest in the areas of explosive sensitivity, and safety and reliability of weapons. The writers would like to acknowledge the assistance of Dr. R. R. Bernecker in this work.

Julius W. Enig

By direction
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INTRODUCTION

Although studies of deflagration-to-detonation transition (DDT) have been underway for some years, the transitional behavior in aluminized explosives has been very little studied. Moreover, relatively few pressed explosives have been examined for DDT at porosities as low as 10%. Those reported are PETN,\(^1\) possibly HMX,\(^2\) and several pure and waxed HE studied in our Laboratory. It is the purpose of this report to present the results of a study of aluminized HMX at 90% theoretical maximum density (TMD). Hence this work supplies much needed data on both aluminized and highly compacted HE.

The composition range was 10-40% Al. Series were prepared with two different particle sizes of Al (5 and 95μ) to study the effect of that parameter on the transitional behavior. All mixtures which exhibited transition followed the DDT mechanism of Reference 3 with the exception of the convective flame front. This front was not detected in most mixtures at 90% TMD; however, the mixtures of 75/25 and 70/30 HMX/5μ-Al did show initial front velocities as low as 1.1 mm/μs.

EXPERIMENTAL ARRANGEMENT AND PROCEDURE

The experimental setup and procedures have been described in detail elsewhere.\(^3\),\(^4\) The apparatus consists of a seamless steel

tube with heavy end closures. The column length of the 0.35 g of 25/75 B/KNO₃ ignitor is 6.3 mm; the length of the explosive column is 295.4 mm. Each charge is examined by x-ray prior to its instrumentation.

The DDT tube is instrumented with ionization probes (IP) and strain gages (SG) to monitor ionization fronts and internal pressure, respectively. For brevity, henceforth ionization probes will be referred to as probes; strain gages, as gages. As before, both custom-made and commercial probes are used; distance-time (x-t) data from each are distinguished on the graphs. The number of gage locations for monitoring internal pressure is generally four or five per tube. The gage output is reported in strain (ε) or microstrain (με). In a static calibration of the tube, the gradient is 112 με/kbar up to the elastic limit at 2.2 kbar. From 2 to 4.7 kbar, the microstrain increases from 225 to 788.

One difference in procedure from that reported in Reference 3 is in the determination of the predetonation column length, ℓ. In the case of 91/9 RDX/wax, it was possible to use the intersection of the postconvective wave front with the extrapolated detonation front to locate ℓ in the x-t plane; the value was confirmed by that obtained from tube fragments. Here, and in general for HE other than 91/9 RDX/wax, ℓ is more reliably determined from markings on the tube fragments; it is checked for consistency with the probe and gage records in the x-t plane. Unless otherwise indicated, ℓ values are measured to ± 3 mm.

The HMX used was Class A (5-200μ); it satisfied the relevant military specification and had the identification number X906. Spherical aluminum powder was obtained from the Valley Metallurgical Processing Company and the Naval Ordnance Station, Indian Head, Maryland. The two batches, designated H5 and H95, had the respective nominal particle sizes of 5 and 95μ. The mixing procedure was that used in Reference 4 for one kg batches.

**EXPERIMENTAL RESULTS AND DISCUSSION**

Eleven shots were made: one on 70% TMD HMX, five on the coarse Al mixtures, and five on the fine Al mixtures. Detailed tables, records for each shot, and discussions of them appear in the Appendix. Table 1 contains a summary of these data.

SG records of these shots were examined carefully for any features which consistently differed from those of records from more porous charges. Because of the lower porosity we thought it possible that these charges might transmit more stress to the tube walls through the solid explosive rather than from gas pressure on the walls. However, no distinctive difference was found in these SG records as compared to those obtained on charges of TMD ≤ 70.
Table 1. Summary of Data on Effect of Aluminum on DDT of HMX

<table>
<thead>
<tr>
<th>Shot No.</th>
<th>%Al</th>
<th>$\rho_o^3$ g/cm$^3$</th>
<th>$\Theta$TMD</th>
<th>Al/O</th>
<th>$l$ mm</th>
<th>$29\Delta t_D$ $\mu$s</th>
<th>$29\Delta t_E$ $\mu$s</th>
<th>D mm/μs</th>
</tr>
</thead>
<tbody>
<tr>
<td>105C</td>
<td>0</td>
<td>1.32</td>
<td>69.6</td>
<td>0.0</td>
<td>41</td>
<td>2</td>
<td>12</td>
<td>7.0</td>
</tr>
<tr>
<td>102C</td>
<td>10</td>
<td>1.76</td>
<td>89.7</td>
<td>0.153</td>
<td>105.0</td>
<td>25.6</td>
<td>42.6</td>
<td>8.3</td>
</tr>
<tr>
<td>113C</td>
<td>10</td>
<td>1.77</td>
<td>90.4</td>
<td>0.153</td>
<td>47.8</td>
<td>8.5</td>
<td>9.5</td>
<td>8.3</td>
</tr>
<tr>
<td>101C</td>
<td>20</td>
<td>1.82</td>
<td>90.2</td>
<td>0.343</td>
<td>56.6</td>
<td>9.4</td>
<td>9.4</td>
<td>8.3</td>
</tr>
<tr>
<td>103C</td>
<td>30</td>
<td>1.86</td>
<td>89.1</td>
<td>0.588</td>
<td>78.2</td>
<td>25.3</td>
<td>26.1</td>
<td>8.0</td>
</tr>
<tr>
<td>108C</td>
<td>40</td>
<td>1.94</td>
<td>90.0</td>
<td>0.915</td>
<td>59.2</td>
<td>17.8</td>
<td>17.8</td>
<td>7.7</td>
</tr>
</tbody>
</table>

**H95 Al**

| 111C    | 10  | 1.78                | 90.9        | 0.153| 56.4   | 10.8           | 16             | 8.0    |
| 107C    | 20  | 1.83                | 90.5        | 0.343| 92.2   | 31.0           | 35.8           | 8.0    |
| 112C    | 25  | 1.85                | 90.4        | 0.458| 149.4  | 74.0           | 60.5           | 8.2    |
| 110C    | 30  | 1.89                | 90.4        | 0.588| 295$^d$| 120.0$^d$      | 106.5$^d$      | b      |
| 106C    | 40  | 1.96                | 90.8        | 0.915| -      | -              | -              | -      |

a. $29\Delta t_D$ is the relative time to detonation referred to the time of discharge of an IP at $x = 29$ mm. Similarly, $29\Delta t_E$ is the relative time from the arrival of a compression wave at $x = 29$ mm to the onset of detonation.

b. Charge failed to show steady state detonation, but was rapidly approaching it. See Figure A10a. In view of the velocity data, a large dent in the closure bolt, and the lack of a clear demarkation on the tube wall, it has been assumed that onset occurred at the very end of tube.

c. All instrumentation failed on this shot, but it was clear from the tube fragmentation that transition had not occurred. See Figure A11.

d. Estimate and lower limit of actual value.

e. Correction from $\Delta t_D$ to $\Delta t_E$ appeared negligible because of the location of first compression front.
In 90% TMD 91/9 RDX/wax, the first compressive front, as outlined by SG data, occurred behind the first ionized front (IP detection) which travelled at about 0.6 mm/μs. In 90% TMD 94/6 RDX/wax, the first compressive front occurred ahead of the IP front which travelled at about 1 mm/μs. The 90% TMD aluminized HMX mixtures gave space-time diagrams resembling those of the corresponding 94/6 RDX/wax mixture. The first front detected by the IPs had velocities ranging from 1.1 (at high Al contents) to about 3 mm/μs. These values are too high for convective fronts; they are probably associated with pressure induced reactions.

Figure 1 shows the space-time diagram for 75/25 HMX/5μ-Al. It is probably the best record of the series because the high aluminum content has spread out the process so that more features can be resolved. We believe it shows the typical DDT behavior in all the aluminized mixtures although the features are not as well resolved in other records. Note that the rate of discharge of the IPs is accelerating from a velocity of 1.1 mm/μs. The first detected compression front starts slightly later and in the region near the ignitor; it travels at 2.2 mm/μs (a reasonable sound speed for this material), rapidly overtakes and passes the IP front which then speeds up to travel at about 2.1 mm/μs. The reaction responsible for the initial acceleration of the IP front has evidently occurred before the first IP responds. Although a convective flame front was not detected, the transition seems to follow the DDT model originally proposed, i.e., accelerated burning leading to a series of compressive waves which coalesce to form a shock of sufficient strength to initiate detonation.

Figures 2-4 show the effect of aluminum particle size and content on the DDT parameters, predetonation column length (L), relative time to detonation (AtD), and the interval between the formation of the first compressive front and the onset of detonation (AtE). The relative times (See Table 1) have been taken for a probe positioned at x = 29 mm; the first probe was located at 22 or 29 mm for these shots rather than at the more customary 41 mm. These figures show quite clearly that all three DDT parameters increase with increasing %Al and that the fine Al is much more effective in causing the increase than is the coarse.

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*Note that AtD is determined entirely by IP measurements whereas AtE is determined chiefly by SG data.
FIGURE 1  SHOT 112C ON 90.3% TMD 75/25 HMX/Al, \( \rho_0 = 1.85 \text{ g/cm}^3 \) DISTANCE-TIME DATA.
( ○ WOL PROBES, ● COMMERCIAL PROBES, □ SG EXCURSION TIME)
SEE ALSO FIGURE A9.
The estimated value for the mixture containing 30% fine Al is a lower limit. In other words, the experimental result was $l > 295$ mm. Hence all data points marked with an asterisk are estimates that might well be lower than the actual value. The Appendix discusses this and the scatter of the data for the mixtures containing coarse Al. In the latter case, two obvious factors might contribute: (1) the mixing procedure seemed less effective with the coarse Al than the fine, i.e., the final mixtures appeared less uniform; and (2) the compaction process during charge preparation had produced a number of demonstrably poor charges. X-ray examination through a thick-walled, steel tube is, however, an insensitive measure of charge uniformity, and might easily fail to detect flaws which could influence DDT. Thus data from one of the charges containing 10% 95μ Al (No. 102C) were omitted when fitting the curves of Figures 2-4 because they contradict the trend for the effect of increasing Al shown by both coarse and fine Al. We believe that the result for 40% Al is low and that for 30% Al about right. However, since the curves can be, and have been, drawn between these two points without changing the qualitative trend, repeat shots did not seem justified.

In a recent report it was shown that waxed Class A RDX and waxed Class A HMX fall on the same curve, $l$ vs %wax, when both charges are at 70% TMD. Assuming that equivalent behavior is also evident at 90% TMD, we can use the data for 91/9 RDX/wax and 94/6 RDX/wax at 90% TMD to compare the effects of Al and wax on HMX. In this way we find that 20-21% 5μ-Al and > 30% 95μ-Al in HMX have about the same values of $l$, $\Delta t_D$, and $\Delta t_E$ as 6% wax in HMX. Similarly, 25% (or more) 5μ-Al results in $l$ and $\Delta t$ values close to those for 9% wax. In other words, on a weight basis, wax is far more effective in slowing down the transition than is Al. It is also more effective on a volume basis although the difference is not great.

In Figures 2-4, curves for both fine and coarse Al extrapolate to about the same value for 0% Al, or for 90% TMD HMX, as they should. A 90% TMD charge of HMX was not run because we did not want it prepared on the hydraulic press at this time. The extrapolated values were not very different from the experimental value of 70% TMD HMX which was run to check the new batch of HMX. The latter value is shown on the figures.

Both from the shape of the curves and parallel knowledge of waxed RDX, a good correlation between $\Delta t_E$ and $l$ would be expected. This is indeed the case as Figure 5 shows; moreover the scatter of Figures 2-4 has been eliminated. If the one estimated point is too low, the curve may be linear as was the case for a smaller range in the waxed RDX.
**FIGURE 2** PREDETONATION COLUMN LENGTH VS ALUMINUM CONTENT FOR 90% TMD HMX/Al. ○ 5μAl; □ 95μAl; ◇ 70% TMD HMX; *ESTIMATED VALUE.

**FIGURE 3** RELATIVE TIME TO DETONATION VS ALUMINUM CONTENT FOR 90% TMD HMX/Al. ○ 5μAl; □ 95μAl; ◇ 70% TMD HMX; *ESTIMATED VALUE.
The curve of Figure 5 suggests that $\Delta t_E$ is a function of $l$ and independent of both Al particle size $\delta$ and aluminum concentration. For the waxed RDX and the waxed HMX series at 70% TMD, over the range of $l < 140$ mm, Class A RDX, Class A HMX, and 115$\mu$-HMX all fell on the same curve, $\Delta t_E$ vs $l$, within experimental error. However, data at $l > 140$ mm indicated separate curves for the two HMXs and possibly for the RDX as well; the difference was small and became smaller as $l$ decreased.* Hence the HMX particle size effect on the relation is negligible at $l < 140$ and small (if significant) at $l > 140$. It is not inconsistent with a negligible effect of Al particle size, in aluminized series, as shown in Figure 5.

In all of the waxed HE, data points appeared on the $\Delta t_E$ vs $l$ curve in order of increasing $l$ which corresponded to the order of increasing diluent. This is also true of the fine Al series shown in Figure 5. For the coarse Al series, there was much scatter (Figures 2-4). Hence these points (squares in Figure 5) do not appear in the order of increasing %Al, but are in a random order. Their location near the curve might be fortuitous.

At 90% TMD, aluminum would not be expected to have much effect on the detonation velocity. For about 30% Al in RDX, the decrease in $D$ should be about 5% or 0.4 mm/ps. Table 1 shows a decrease of 0.3 mm/ps for this concentration of coarse Al and none (or an increase) for the fine. The size of the effect is just about that to be expected from experimental error so that it cannot be considered of much significance.

Finally, it is of interest that the mixtures with coarse aluminum caused greater fragmentation damage (smaller fragments and more of them) than did mixtures with fine aluminum. The former behaved more like the pure explosive than the latter in both DDT and fragmentation.

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*On the other hand, waxed RDX had distinctly different $\Delta t_E$ vs $l$ curves at 70% and 85% TMD; this difference is large and it is also one reason that series of constant composition HE with varying degrees of compaction fail to show a correlation between $\Delta t_E$ and $l$. 
SUMMARY AND CONCLUSIONS

1. Addition of fine (5μ) and coarse (95μ) aluminum to HMX, in charges at 90% TMD, increases predetonation column length (l) and the relative times to detonation (∆t_D, ∆t_E).

2. All three DDT parameters increase with increasing %Al. They are interdependent and a good correlation exists between ∆t_E and l.

3. Al causes only a small decrease in detonation velocity D at 90% TMD.

4. Strain gage records from the 90% TMD charges showed no features different from those previously obtained on charges of higher porosity.
REFERENCES


APPENDIX A

DETAILED DATA FOR ALL SHOTS

Measurements for all shots are given in Table A1. The distance-time plot and the strain-time plot, where available, are displayed in Figs. A1-A10 as parts a and b, respectively. Fig. A11 shows the effect of Al concentration on tube fragmentation, and supplements Figs. A7-A10 for 5µ Al.

HMX

Fig. A1 displays the data for the current batch of HMX of 70% TMD. Its short accelerating convective flame front (0.5 mm/µs) followed and then preceded by a compressive front (at 1.6 mm/µs) compares very well, i.e., almost reproduces, the data for 70% TMD RDX and HMX given in reference 5. There RDX had an initial front velocity of 0.5 mm/µs followed by PC front at 1.3-1.5 mm/µs (see Ref. 5, Fig. 4). Similarly HMX (Ref. 5, Fig. D1) exhibited analogous velocities of 0.4 mm/µs and 1.3 mm/µs. In each case, the compressive wave originated behind the convective front and near the ignitor/explosive interface. It then overtook and passed the convective front. The apparent DDT mechanism was in each case that proposed in Reference 3. Evidently Batch X906 shows behavior equivalent to that of previous batches of Class A RDX and HMX.

Mixtures with 95µ-Al

Mixtures with the coarser aluminum were harder to prepare, i.e., they seemed less uniform, than those made with the finer aluminum. Their records were also somewhat more difficult to read and interpret.

Fig. A2 shows data from the first shot on 90/10 HMX/Al. Although much of the SG records was obscured by interaction between the IP discharges and the SG, two compression fronts were observed: one preceding the front detected by IPs (the common order found in this study) and one almost coincident with it. See Fig. A2a. The velocities of all three fronts were much the same, and, at 3 mm/µs, higher than the estimated sound velocity of the original charge.

Fig. A3 displays data from the shot of a second 90/10 HMX/Al charge. The SG records are cleaner, but because of the very short predetonation column length, only two pressure excursions were detected: one ahead of the front outlined by IPs (the usual location) and one obviously responding to the detonation. See Fig. 3Aa. Again the initial IP front has a velocity (2.7 mm/µs) to be expected of a compressive wave, and the discharge of the IPs is attributed to a pressure induced reaction.

Although these charges show satisfactory replication in the detonation region, they are thoroughly unsatisfactory in reproducing the predetonation column length L. The cause of the different predetonation behavior is attributed to differences in charge preparation not detectable by x-ray examination and possibly to...
differences in uniformity of the mixture as well. Predetonation phenomena of non-steady combustion would be far more sensitive to small physical differences in the charge than would detonation.

Data from the 80/20 HMX/Al charge are shown in Fig. A4. This is the first case in which the pressure excursion is shown as squares connected by a line. Here the squares indicate the boundary of a region within which the excursion occurred. As Fig. A4b shows, limits rather than a single point are required here because (1) interaction of IPs with SG obscures portions of the record and (2) irregularities appear in the record of SGs located at or near the point of detonation onset. The last two excursions are the result of the detonation; in fact, the next to the last SG (x=56.8 mm) is at the point of onset of detonation (x=56.6 mm). The first two excursions show a pattern of the compression wave originating behind the front outlined by IP response; this front is accelerating and travelling at a higher velocity so that it overtakes the IP front. A similar pattern was observed in the higher % TMD waxed explosives (5), and is probably not detected in most of the aluminized HE because the triggering IP responds too late. The presence of Al probably lowers the temperature and hence the concentration of charged particles below the level necessary for probe response.

The Fig. A5 for 70/30 HMX/Al, the compression wave (v3 mm/μs) has been drawn in a pattern similar to that of Fig. A4. However, two parallel compression fronts sandwiching the IP front could have been drawn through the data from the first three SGs. The record for the last, which happened to be at the point of detonation onset, clearly shows the dip in the strain-time record associated with the passage of a strong shock (See Fig. A5b). The corresponding pressure excursion therefore belongs where it falls in Fig. A5a at the beginning of the detonation region.

In Shot 108C on 60/40 HMX/Al, the SG records were lost. Hence Fig. A6 consists of only the distance-time diagram with no compression fronts shown. The initial velocity, obtained from the IPs is 1.4 mm/μs, the lowest observed in this series.

Mixtures with 5wAl

From Fig. A7 on, the data displayed are those obtained from mixtures of HMX with fine aluminum. Very few new features appear in these shots, but their results do permit some generalizations. For example, all the IP fronts are of somewhat lower velocity than those of the comparable mixture with coarse Al.

Fig. A7 for 90/10 HMX/Al needs no special comment except that the response of the SG at x=1 is exactly what might be expected for the passage of a shock (in this case, a detonation): a rapid rise in strain followed by a fairly rapid decay and gage destruction. It was pointed out in several earlier records, more often we observe a sharp dip rather than a sharp rise in the presence of a shock.
Thus Fig. A8b at the SG just ahead of \(x=l\) shows a sharp dip in the data from 80/20 HMX/Al. However the SG just beyond \(x=l\) shows two dips followed by a very large and rapid rise.

Fig. A9a for 75/25 HMX/Al shows a better example of the same pattern noted in Fig. A4a, that which every charge might show with earlier IP response. Here \(l=149\) mm and the SG at 118 mm shows a very sharp rise, followed by decay; that at 143 mm shows only minor disturbances until its final rapid rise.

Fig. A10 for 70/30 HMX/Al shows that this charge may not have detonated before the end of the tube. However, the displacement of the IP datum at 117 mm and the monotonic increase in front velocity thereafter indicate that the detonation velocity (\(\sim 8\) mm/\(\mu s\)) might have been achieved just at the end of the tube. That plus the fact that the closure bolt was heavily dented seem to justify an estimate of \(l\sim 295\) mm. This cannot, of course, be justified by tube markings.

All records were lost for the shot on 60/40 HMX/Al, but from data on the previous shot, lack of tube marking, and type of fragments, this shot certainly did not reach detonation although there was a dent in the closure bolt. Fig. A11 shows fragments from Shots on 90/10 and 75/25 HMX/Al (which detonated) as well as for 70/30 which may just have reached detonation and for 60/40 HMX/Al which failed. It is quite evident that fragmentation efficiency decreased as aluminum content increased whether or not a detonation occurred.
TABLE A1
DETAILED DATA FOR SHOTS

<table>
<thead>
<tr>
<th>Shot No.</th>
<th>109C</th>
<th>102C</th>
<th>113C</th>
</tr>
</thead>
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<tr>
<td><strong>HE</strong></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>HMX</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>$\rho_0 (\rho_v)$, g/cm$^3$</td>
<td>1.322(1.900)</td>
<td>1.756(1.958)</td>
<td>1.771(1.958)</td>
</tr>
<tr>
<td>% TMD</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td>69.6</td>
<td>89.7</td>
<td>90.4</td>
</tr>
<tr>
<td><strong>IP Data</strong></td>
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<td>$t$</td>
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<tr>
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* Custom-made probe.

a. All HMX is Class I(A) from Batch X-906.
b. Increase in (dp/dt) or abrupt termination of record.
c. H95 Al
d. H-5 Al
e. F is failure to achieve steady state detonation.
f. Uncorrected baseline shift on these records.

Units of $x$ are mm, of $t$, \textmu s.
## TABLE A1 (Cont'd)

### DETAILED DATA FOR SHOTS

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<td>60/40 HMX/A1&lt;sup&gt;c&lt;/sup&gt;</td>
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<td>$\rho_o(\rho_v)$, g/cm&lt;sup&gt;3&lt;/sup&gt;</td>
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<td>1.859(2.066)</td>
<td>1.940(2.155)</td>
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* Custom-made probe
** Estimated

- c. H95 A1
- f. Uncorrected baseline shift on these records.
TABLE A1 (Cont'd)

DETAILED DATA FOR SHOTS

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**IP Data**

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\( \lambda (\text{mm}) \)  
56.4  
92.2  
149.4

\( 29\Delta t_D(\mu s) \)  
10.8  
31.0  
74.0

\( 29\Delta t_E(\mu s) \)  
16.0  
35.8  
60.5

\( D(\text{mm/\mu s}) \)  
8.0  
8.0  
8.2

\( \text{pre} D(\text{mm/\mu s}) \)  
2.2  
1.6x2.1  
1.1-2.0x2.2

* Custom-made probe

d. H-5 Al.
f. Uncorrected baseline shift on these records.
TABLE A1 (Cont'd)

DETAILED DATA FOR SHOTS

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<td>60/40 HMX/Al&lt;sup&gt;d&lt;/sup&gt;</td>
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**IP Data**

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**SG Data**

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| $\lambda$(mm) | > 249(295**) |
| $29\Delta t_D$(µs) | > 114(120**) |
| $29\Delta t_E$(µs) | (120**) |
| D(mm/µs) | $F^e$ |
| pre D(mm/µs) | 1.1 |

---

* Custom-made probe
** Estimated

<sup>d</sup> H-5 Al.
<sup>e</sup> F is failure to achieve steady state detonation.
<sup>f</sup> Uncorrected baseline shift on these records.
a. Distance-time data. (O WOL probes, ● commercial probes, □ SG excursion time).

b. Strain-time data. (Each curve, except the lowest, has been raised 200με — or an integral multiple thereof — for better data display.)

FIGURE A1 SHOT 109C ON 69.5% TMD HMX, ρ_o = 1.32 g/cm³.
FIGURE A2  SHOT 102C ON 89.5% TMD 90/10 HMX/95μ-Al, ρ₀ = 1.76 g/cm³. (KEYS OF FIGURE A1)
FIGURE A3  SHOT 113C ON 90.3% TMD 90/10 HMX/95μ-Al, $\rho_0 = 1.77$ g/cm$^3$. (KEYS OF FIGURE A1)
FIGURE A4 SHOT 101C ON 90.2% TMD 80/20 HMX/95µ-Al, \( \rho_o = 1.82 \text{ g/cm}^3 \). (KEYS OF FIGURE A1)
FIGURE A5  SHOT 103C ON 89.0% TMD 70/30 HMX/95μ-Al, \( \rho_o = 1.86 \text{ g/cm}^3 \). (KEYS OF FIGURE A1)
FIGURE A6: DISTANCE-TIME DATA FOR SHOT 108C ON 89.9% TMD 60/40 HMX/95µ-Al, $\rho_0 = 1.94 \text{ g/cm}^3$. (KEYS OF FIGURE A1)
FIGURE A7  SHOT 111C ON 88.9% TMD 90/10 HMX/5μ-Al, ρ_o = 1.78 g/cm³. (KEYS OF FIGURE A1)
FIGURE A8  SHOT 107C ON 90.4% TMD 80/20 HMX/5µ-Al, \( \rho_o = 1.83 \text{ g/cm}^3 \). (KEYS OF FIGURE A1)
FIGURE A9  SHOT 112C ON 90.3% TMD 75/25 HMX/5μ-Al, ρ_o = 1.85 g/cm^3. (KEYS OF FIGURE A1)
FIGURE A10  SHOT 110C ON 90.3% TMD 70/30 HMX/5\mu-Al, $\rho_o = 1.89$ g/cm$^3$. (KEYS OF FIGURE A1)
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