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EXTENDED HEATING ABLATION OF CARBON PHENOLIC AND SILICA PHENOLIC

R. W. Farmer

Air Force Materials Laboratory Wright-Patterson Air Force Base, Ohio

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CARBON PHENOLIC AND SILICA PHENOLIC

R. W. Farmer

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FOREWORD

This report was prepared by the Thermally Protective Materials Section, Composites and Fibrous Materials Branch, and was initiated under Project 7340, "Nonmetallic Composites and Materials," Task 734001, "Thermally Protective Plastics and Composites." The work was administered under the direction of the Nonmetallic Material: Jivision, Air Force Materials Laboratory (AFML). The AFML project engineer was Mr. R. Farmer (AFML/MBC). The effort covered the period of September 1972 through January 1973.

This report summarizes and acknowledges an independent, intercomparative, and in-depth analysis of a large body of information resulting from two individual investigations of extended heating ablation. The original work, which includes engineering details as well as empirical observations requiring subjective interpretations, consists of the following:

"Ablative Materials Characterization. Part III. Turbulent High Heating Loads, High Pressure Tests, and High-Modulus Fibrous Composites," B. J. Mitchel and P. J. Roy. AFML-TR-69-188, Part III, December 1970. AF Contract F33615-68-C-1425, Avco Government Products Group.

"Ablative Materials for High Heat Loads. Part I. Environmental Simulation and Materials Characterization," P. W. Juneau, Jr., J. Metzger, L. Carkowitz, and F. P. Curtis. AFML-TR-70-95, Part I, June 1970. AF Contract F33615-69-0-1503, General Electric Company.

The reader is referred to these reports (or to their authors) for acquisition of additional specific technical details. This report was submitted by the author February 1974. This technical report has been reviewed and is approved.

T. J. REINHART, JR., CHIEF Composite and Fibrous Materials Branch Normetallic Materials Division Air Force Materials Laboratory

ii

ABSTRACT

An analysis was made of experimental and analytical investigations of the ablation of carbon phenolic and silica phenolic composites under extended heating conditions. Specimens of up to 8.75 sq. in. in area and instrumented with indepth thermocouples were characterized under stepwise pulses of either five minutes (2 steps) or up to 1.4 minutes (to 5 steps) in duration using two air arc heaters. The nominal peak heat load was 35,000 Btu/sq ft. Internal and surface temperatures, recession rates, and recession patterns in the residual char were not anomalous for the two step, low shear (to 2.5 lb/sq ft) runs. Charringablator theory indepth and surface temperature responses agreed well with experimental results for a carbon phenolic. For the five step condition with a moderate peak shear (30 lb/sq ft) there was cine film evidence of micromechanical surface removal at late times. Micromechanical effects, by difference, were further consistent with theory. Reliable composite properties were found to be necessary to accurately model extended heating ablation.

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TABLE OF CONTENTS

SECTION		PAGE
I	INTRODUCTION	1
II	EXPERIMENTAL MATERIALS	4
III	ABLATIVE CHAPACTERIZATIONS	8
	1. Air Arc heaters	8
	2. Specimen Measurements	16
IV	COMPUTER CODES	18
٧	RESULTS AND DISCUSSION	21
	1. Two Step Exposures	21
	2. Multiple Step Exposures	37
٧I	SUMMARY AND CC +OLUSIONS	4;
VII	RECOMMENDATIONS FOR FUTURE STUDY	45

ILLUSTRATIONS

1

FIGURE		PAGE
1.	Extended Heating Characterization Environments	2
2.	Tandem Electrode Air Arc Heater	9
3.	Arc Heater And Specimen Arrangement	10
4.	Arc Heater Installation	11
5.	Multiple Electrode Air Arc Heater	13
6.	Arc Nczzle And Specimen Arrangement	14
7.	Arc Heater Installation And Instrumentation	15
8.	Char Depth And Surface Recession For Carbon Phenolics	23
9.	Internal Temperature Histories For R2 Carbon Phenolic	24
10.	Internal Temperature Histories For R3 Carbon Phenolic	25
11.	Surface Temperature Histories For Carbon Phenolics	26
12.	Char Depth And Surface Recession For Rl Silica Phenolic	29
13.	Internal Temperature Histories For Rl Silica Phenolic	31
14.	Surface Temperature History For Carbon Phenolic (R3) And Silica Phenolic (R1)	32
15.	Predicted Internal Temperature History For Carbon Phenolic	34
16.	Internal And Surface Temperature Histories For R6 Carbon Phenolic	39
17.	Surface Recession For R6 Carbon Phenolic	42

vi

AFML-TR-74

TABLES

TABLE		PAGE
I	Nominal Environmental Parameters	3
II	Composite Fabricational Parameters	5
III	Mean Thermocouple Depths	7
IV	Representative Ablation And Composite Properties	19
۷	Specimen Dimensional Changes	22
VI	Experimental Results - Two Step Exposures	28
VTT	Experimental Results - R6 C/P Multiple Step Exposures	38

vii

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SECTION I

INTRODUCTION

Extended heating periods at moderate convective heat fluxes represents a relatively new aerospace environment for efficient charring ablative materials. When environmental regimes with an early thermal soak followed by a second lengthy period became of interest, there was little applicable data for carbon phenolic and silica phenolic. This resultant study was oriented toward definition of any anomalous phenomenology. One postulated case, for example, was the unpredictable formation of relatively weak char regions during early heating with late thermomechanically-induced particle loss associated with time-dependent variables as heating rate, pressure, and shear.

As illustrated by Figure 1, two heating pulses were selected to bracket potentially critical response modes. One condition consisted of a long, low heat flux period followed by a step change to a moderately high level. The second environment involved up to five steps with a rapid increase and then a decrease in heat flux. The rominal peak heat load was 35,000 Btu/sq ft for both conditions. The peak th ar stress was about 12 times larger for the second case as compared to the first one. Table I summarizes nominal values of additional environmental parameters.

The composite specimens, with a cloth layup angle of about 20°, were supported against the rectangular wall of the air arc heater nozzle. The major measurements consisted of internal temperature history, chardepth, surface recession, the surface temperature history, and weight loss. The experimental response of the carbon phenolics was compared with the predictions of two transient charring-ablator computer codes.

AFML-TR-74-45



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N.MINAL ENVIRORMENTAL PARAMETERS

	Two St	en		Multi	ple Step		
Item/Step Number		2	-	8	ŝ	4	Ś
Enthalyy, Btu/lb Heat Flux, Etu/sq ft-sec	5000 25	5000 1440	1050 220	3500 470	6000 660	3850 510	175(290
Fressure, atm Plenum Specimen		-+ +-	0.0	ج. د.،	5.2	5	
Shear Stress, lb _f /sq ft Time Interval, sec	0.4 230	2.5	88	8 5 7	85	0 2 2 2	85
Total Time, sec Total Heat Load, KBtu/sq ft	230	300 35 . 6	20 4.4	35 11 . 4	48 20 . 0	64 28.2	ઝુ.સુ.

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AFML-TR-74-45



SECTION II

EXPERIMENTAL MATERIALS

Ablative heatshields are frequently prepared by wrapping a prepreg tabe on a mandrel under tension at a layup angle near 20°. The tape is made by impregnating bias cut cloth with a phenol.c varnish and thermally advancing the resin cure. The mandrel assembly is cured to completion by autoclave or other suitable heating methods under pressure. The complex, biangular configuration of the cloth fibers in a cross-sectional element of the tapewrapped composite may be important in ablative response.

Three composites, codes R1, R2, and w3 were prepared using laboratory autoclave fabrication. Cloth strips cut on a 45° bias angle were impregnated with a phenolic varnish and tacked + an aluminum fixture. An approximate 20° layup angle was maintained with the aid of an end support. The layup was vacuum bagged, thermally edvanced, and transferred t. an autoclave, cured, removed from the vacuum bag, and postcured. The R6 composite was prepared by high pressure compression molding to achieve maximum density for a thick section. A 20° layup angle was approximated in machining the specimen; a 45° fiber bias angle was not simulated. Table II summarizes fabricational parameters for these composites.

R1, R2, and R3 composite sections were machined for two step exposures. The nominal dimensions were 5 inches in length by 0.5 inches in thickness by 1.75 inches in width. The specimen was prepared by bonding on a 2024-T3 aluminum substructure (chromic acid etched) 0.0625 inches in thickness with HT-435 epoxy-phenolic film adhesive. The composite/plate assembly was clamped for curing for one hour at 350°F. The specimen was instrumented with five 36 AWG Chromel alumel thermocouples and a tungsten reference wire. The thermocouple holes, prepared by axial drilling, were located parallel to the heated surface to minimize heat losses as well as being staggered by about 0.006 inches along the lateral centerline to minimize any interference.

TABLE II

COMPOSITE FABRICATIONAL PARAMETERS

constituents ^a /Code	R1	R2	R3	R6 ^e
Retrforcement	Refrasil	Carbon	Carbon	Carbon
Cioth Tvna	C-100-43	CCA1-1541	Pluton B1 HP	Pluton B1 HP
Docin Trme	1155 95	11SP 55	DP 25-10	SC1 UC3
Ducate Time	EX 5020	FX SCSSA	MXC-31HP	MXC 31 -HP
richics ijre Mannard				
Lepter	17.0	16.1	11.1	6.8
Resin Content. 8	27-33	34-36	45-50	44.5
Resin Solids.		•		40-45
Volatiles. &	4.0	4.1	6.4	<i>ن</i> • 5
Jurecia				
Pressure. psi	150	150	150	1000
Time (hr) @ Each	2,200°;2,250°;	Same as R1	Same as R1	l hr @ 25% incre-
Temperature (OF)	4,3250			ments, 175° to 350°
Composites			-	(
Density, gm/cc	1.67	1.51	1.43	1.42
Resin Content. %	36.8	35.0	48.5	43.2
Size, inch	12x12x0.60	12x12x0.50	12x12x0.48	6x6x4
^a C-100-48, CCA1-1641:	HITCO., Fluton Bl	HP: 3M Co., DP	7-10: Ironsides	Resins Co., mitter

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SC1008: Monsanto Chemical Co., USP 95, FM Prepress: US Polymeric, Inc., MXC Prepress: Fiberite Corp., ^bRl, R2, R3: staged @ 15 psi for 1 hr @ 180°F in a vacuum bag. CRl, R2, R3: autoclave cure for R1, in a vacuum bag. R6: compression molded to stops, colled under pressure overnight. Postcure for R1, R2, R3: removed from vacuum bag, 4 hr @ 400° F. Postcure for R6: 16 hr @ 200° . 2 hr 3° 225°. 2 hr 2° 250°, 4 hr @ 275° , 16 hr @ 300° , 4 hr @ 325° . 4 hr @ 750° F, 600° Cooled in oven to 150°F. eReinforce-fort, resin, and prepreg produced to meet Fiberite Specifications M72201, M74102, and M72251 Rev B high-purity specifications, respectively.

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AFML-TR-74-45

The tungsten wire was located near the specimen center. As summarized by Table III, the nominal thermocouple depths were 0.1, 0.2, 0.3. 0.4, and 0.5 inches, the latter thermocouple being located below the composite in the bond region. The actual depths were estimated by sectioning exposed specimens.

The R6 composite was machined for multiple step exposures. The nominal dimensions for the specimen were 2.50 inches in length by 0.75 inches in thickness by 1.00 inches in width. The specimen was instrumented with four tungsten-rhenium high temperature or Chromel alumel thermocouples at depths consistent with the estimated response of a specimen for a particular exposure sequence. The holes were axially drilled and the depths of the holes were estimated by using a dowel pin as a probe. The thermocouple/sheath assembly was bonded into place. The locations were staggered axially and laterally to minimize any edge heat loss, or mutual interference effects. There were fewer thermocouples near the surface for those specimens intended for a larger rumber of steps (Table III). TABLE III

MEAN THERMOCOUPLE DEPTHS*

Ś	644.0		
4	404° 0		0.226 0.276 0.329 0.376 0.376
ŝ	0.291		0.176 0.226 0.279 0.335 0.335
2	0.200		0.124 0.222 0.223 0.173 0.272
÷	0.104		0.076 0.121 0.173 0.174 0.174 0.222 0.224
Run Type/(T/C) Number	Trio Steps	Multiple Steps	

*Depths are in inches from the original surface For two step runs, 36 AWG Chromel alumel tnermo-couples. For multiple step runs, the numbers correspond to the exposure sequence. W+5% Re/W+25% Re thermocouples (beryllia sheaths) to the left of the stepped line; Chromel alumel (alumina sheaths) to the right.

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AFML-TR-74-45

SECTION III

ABLATIVE CHARACTERIZATIONS

1. AIR ARC HEATERS

The arc heater for the two step runs was of the Tandem Gerdien design. As illustrated by Figure 2, air was injected into the two swirl chambers. Most of the air passed through a vortex stabilized arc column, into the plenum chamber, and exhausted through a complex nozzle assembly. The rest of the air was bypassed over the electrodes to remove contaminants and exhausted from the rear of the arc chamber. Two DC generators, connected in parallel, were operated near the maximum rating of 2000 amperes at 1300 volts.

Two step operation was possible by using the .ssembly illustrated by Figures 3 and 4. A bypass port, which exhaus ad most of the effluent during the first step of the exposure, was closed by c graphite plug to obtain the more severe second step. The graphite flow deflector was actuated by a reciprocating air cylinder. The variable nozzle, 0.1104 square inches in throat area, provided near supersonic flow with little fluctuation in the arc plenum pressure after closure of the bypass port. Figure 3 further illustrates the variable nozzle instrumentation, specimen installation, and a graphite insulator used to isolate the specimen from the water-cooled copper hardware.

The effluent bulk enthalpy was estimated by an energy balance. Heat flux calibration measurements were made with a water-cooled calorimeter in the specimen position. A heat flux transducer in the variable nozzle, intended for monitoring during the run, was not satisfactory. Therefore, specimen values were estimated using calibration data and the environmental parameters. Pressure histories were measured for the arc and entension plenums and at three stations above the specimen along the variable nozzle wall. Shear stress was nominally evaluated using Reynold's Analogy. Additional arc operational measurements primarily involved the air mass flow rate, cooling water flow rate and temperature



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Figure 2. Tandem Electrode Air Arc Heater

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AFML-TR-74-45



Figure 3. Arc Heater And Specimen Arrangement

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Figure 4. Arc Heater Installation

change, current, and voltage. The continuously recorded parameters generally stabilized at a constant level within a few seconds after run initiation.

Figure 4 is an overall view of the arc, specimen installation, and supporting instrumentation.

The facility for the multiple step exposures essentially consisted of five air arc heaters discharging into a spherical plenum (Figure 5). The effluent exhausted in a direction normal to the four equally spaced arc heads and parallel to the fifth. The power supply consisted of 2080 heavy-duty, 12-volt storage batteries.

The air mass flow rate and the power to each arc head were programmed both sequentially to step change anthalpy and heat flux as well as separately to maintain a nearly constant plenum pressure with proper effluent expansion. The contoured, rectangular nozzle was 1.405 inches in height by 1.000 inch in width at the exit plane. The nomina) Mach Number was 1.79. Figure 6 illustrates the specimen installation at a 10° inclination angle. The nozzle and specimen holder was made from copper. Figure 7 is an overall view of the source, specimen installation, and supporting instrumentation.

The effluent bulk enthalpy was estimated from a correlation for sonic mass flow at the mozzle inriat, as derived from isentropic flow relationships. The correlation was confirmed by selected enthalpy probe measurements. Heat flux was primarily estimated from correlation curves based upon calibration experiments and theory. The plenum pressure history was recorded and correlated to give a specimen surface pressure. Shear stress was estimated from Reynold's Analogy. The time interval for the steps generally corresponded to a step deflection of an oscillograph record. Steady state current, plenum pressure, and voltage values were generally achieved in less than 0.2 seconds after run or step initiation.

AFML-TR-74-45



Figure 5. Multiple Electrode Air Arc Heater



Figure 6. Arc Nozzle And Specimen Arrangement

AFML-TR-74-45



Figure 7. Arc Heater Installation And Instrumentation

2. SPECIMEN MEASUREMENTS

The specimen measurement procedures were not the same for the two types of characterizations consistent with available apparatus, behavioral differences, and data uses. The measurements during exposure did include an in-depth internal temperature history and a surface temperature history at one station. In addition, the surface recession history was obtained by a photographic technique for the strongly transient multiple step runs. Post-test measurement involved local and/or overall char depth, surface recession, and weight loss.

For two step runs, the in-depth thermocouple responses were recorded by an oscillograph. The surface color temperature was recorded by a two-color pyrometer viewing a nominal 0.25 inch diameter area just outside the nozzle exit. The total specimen weight change was obtained by differential weighing. Axial char depth and surface recession profiles were obtained along the center line of a sectioned specimen using a cathetometer. Time lapse 16 mm color photography of the specimen surface during exposure proved unsatisfactory due to camera malfunctions and other difficulties.

Analysis of the widely variable in-depth thermocouple responses for multiple step runs was facilitated by magnetic tape recording, digitization, and computer processing. An oscillograph was also used for data recording. Brightness temperature and total radiation histories were obtained by an infrared pyrometer and thermopile arrangement, respectively. The reference station was on the centerline 1.5 inches from the nozzle exit. The fields of view were 0.13 and 0.33 inches in diameter for the pyrometer and thermopile, respectively. The surface recession history was obtained at the reference station, 1.5 inches from the nozzle, using 35 mm camera film strips made at a 2 second framing rate. The surface recession was generally not uniform over the specimen. Therefore, a 0.5 inch diameter core was removed from the reference station. This core was used for the assessment of specimen thickness and weight changes. In addition, a mean local areal weight change was

calculated as the difference of the average initial areal weight of the specimen and the areal weight of the core. Color 16 mm movies were made of the specimen surface during exposure.

SECTION IV

COMPUTER CODES

Carbon phenolic response was assessed using two charring-ablator computer codes. While similar in many respects, there were differences relating to formulation, modelling, solution, and property definition. Although descriptions of each were beyond the scope of this report, essential features were reviewed for illustrative purposes.

The transient ablation code for the two step runs modelled finite phenolic resin pyrolysis kinetics, gas phase chemical reactions, and temperature-dependent thermophysical properties. These mechanisms were considered potentially significant for the thick chars found for this environment. The one-dimensional in-depth heat balance included chemical reactions, conduction, gas phase sturage, and solid storage with phenolic pyrolysis kinetics being expressed as an Arrhenius-type correlation. There was zero net conduction at the substructure boundary. The surface energy solution balanced convective heating, pyrolysis gas blockage in the boundary-layer, and surface radiation. Recession rate was represented by a (Knudsen-Langmuir)-type correlation with three empirical coefficients. The governing equations for the code were solved simultaneously at selected in-depth nodes in real time. As the code was terminated, if an incremental solution was not calculated within specific accuracy limits, a modification was necessary to reduce computational time increments in the vicinity of the step change.

The input materials properties were nominal values representative of the R2 composite and typical for this materials class (Table IV). The inputs included both conventionally measured thermophysical properties as well as experimentally measured and estimated data empirically accounting for gas phase chemical reactions and recession rate. The environmental definition inputs included both nominal and adjusted values of the environmental parameters. The adjusted values were based upon an analysis of the experimental data.

	よっ/83	Clace ^B		R6 ^b			R6 Clas	ა წ
	Char Char Zone	Virgin Zone	Char Zone	Virgin Zone	Pyrolysis Gas	Cher Zone	Virgin Zone	Pyrolygis Zone
Density, lb/cu ft Erittance	74.0 0.85	90.4	73.6 0.7	92.0 0.7		68.4	91.2	
Heat of Char Combustion, KBtu/lb			-4.36/-4.60 ^d		-1.8/+3.0 ^d			
Heet of Pyrolysis KEtu/lb				0.48	-	0/0 ⁴	0.205	
Resin Pyrolysis Temperature. ^{OF} Specific Heat, Btu/lb- ^{OF}	0.22 1	to 0.55	0.45	540 0.25	0.80	0.35	1000 0.35	77.0
Specific Heat Temperature Fange, ^{OF}	00 to	4550 ⁰						
Thernal Conductivity, Etu/hr-ft- ^{oF}	0.34 [†]	to 1.60	0.40 to 2.56	0*70		1.0	0.44	
Thermal Conductivity Ferperature Range, ^{OF}	0° to	4550°	540° to 4540°					
^e Representative values f	or R2/1	ໃ3 ບ1a ss	of composites.	b _{Based} ı	upon a nonlin	lear reg	ression a	nalysis of

of steady-state ablation data for R6. ^CTypical experimental or estimated data for R6-type composites. ^dFor 4040°/5840°F, respectively, at] atm pressure. ł ,

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AFML-TR-74-45

TABLE IV

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The code applied to multiple run analysis considered blowing, combustion (gas, solid), the heat of resin pyrolysis, storage (gas, solid), surface radiation, temperature-dependent char thermal conductivity, and thermal conduction in depth (gas, solid). Phenolic pyrolysis was expressed in terms of a step change from virgin material to char at a constant pyrolysis temperature. Surface recession was assumed to result from diffusion-limited oxidation and removal of the char. The rate balance considered blocking, combustion (gas, solid), the oxygen flux, and the pyrolytic gas flux. The equations were solved by a technique well suited to examine the various experimental exposure sequences. Essentially, the relations were transformed into a moving coordinate system, further transformed to increase the grid-point density in regions of high temperature, and formulated into an implicit finite-difference scheme for simultaneous solutions at each step. Small steps were not required when the char front was near the surface.

The input properties for the R6 composite were determined by a nonlinear regression analysis of internal and surface temperature data. This widely used technique was applied to an R6 specimen exposed in a more severe environment to achieve steady-state ablation. These properties were considered adequate for a first assessment in that the maximum difference between the experimental temperatures and temperatures calculated using the code, derived properties, and nominal environmental parameters did not exceed about 200°F at any of four depths or 450°F at the surface. Table IV summarizes the properties obtained from the regression analysis as well as other data measured by various techniques for composites similar to R6 with respect to type of reinforcement and resin. The environmental inputs for the analysis of the response to multiple step heating corresponded to the nominal environmental parameters for that exposure sequence.

SECTION V

RESULTS AND DISCUSSION

1. TWO STEP EXPOSURES

The carbon phenolic composites R2 and R3 were exposed to the two step condition. The approximate mean char thicknesses for R2/R3 were 0.28/0.21 inches and the corresponding mean recession was 0.11/0.06, respectively (Table V). As illustrated by Figure 8, there were differences in the internal char profiles and external surface patterns for these two composites.

There was considerable recession in the leading edge region for R3. An early high recession rate may have been self-aggravating in promoting local environmental changes and a higher than normal recession rate. The excessive ejected material may have further reduced downstream recession. For R2, although the surface was considerably smoother, a higher overall char irregularity was found than for R3.

The char profile unevenness was partially a result of the surface irregularity and porosity. This was due to the station to station measurement technique bias of the char depth data in that the char thickness was taken as the depth difference between the surface and char interface.

There were large differences in the internal temperature responses of R2 and R3 (Figures 9 and 10). R2 provided considerably superior insulative ability during the first step. The temperatures increased rapidly for both specimens after the onset of second step heating.

The surface temperature history consisted of a nearly steady value near 2030°R for the first heating pulse with a step change for the second (Figure 11). The surface temperature for R2 averaged a negligible 120°F higher than for R3 for the first step. While the temperature stabilized near 4070°R for R3 for the second pulse, there was a saddle-like fall-off

		real Wuight <u>os, 1b/Sq.ft</u> Speczmen		0.14 0.95	1.37	2.16	1.44	1.69	2.81	, the norrlo. r load viro and se.
Weight Loss, Em	10.075 15.4% 21.334	A) Core		6.0 1	1.25	2.30	1.38	1.72	3.13	jacent to inud afte u drillir
Initial Naight, E ^m	146.967 133.559 124.970	Recession	Inch	-0.015 0.031	0.083	0.124	0.140	0.145	9.340	he station ad valuos dotorm ractured vhil
Mean Char Thickness, inch	0.200 0.280 0.210	Core Values ^d Weight,	ß	3.481	2.770	2.428	2.672	2.430	1.568	anges oxcludo t tion; residual exit. ^O Core f
Glar Thickness, inch	0,189-0.274 0.171-0.252 0.157-0.231	Discotor.	inch	c. 500	6.501	c. 501	0.502 0 502	0.501	0.501	nd recossion ru ouplo installa rom the nozzlo
Moan Recession, inch	0. <i>0</i> 77 ^b 0.056 0.108	Weight Loss.	E.		10.8	0.1	17.0		22.1	har thíckness a before thirmoc n 1.50 inches f
Recession, inch	0.017-0.102 ^b 0.044-0.087 0.068-0.182	n Values ^c Values ^c	Carlina,	8.77	44.9 45.0	8.77	45.0	1.5.0	14.8	ubstructure. C CInitial values eference statio
Nean Thickness, inch	0.555 0.574 0.558	Specimo	inch inch	0.750	0.750	672.0	0.751	0.750	0.750	and aluminum s 0.014 inches. et centerling r
es ^a - Thickness, inch	0.553-0.570 0.574-0.578 0.550-0.558	posures	lb/cu ft	91.1	91.6 61	5	91.3	91.0	91.1	s adhesive bond thickness was Core drilled
o Step Exposur Composite Type	S/P S/P	ltiple Step Ex	uren step ar Sequence	۲	იი ი 1) (1) - +-	-1-	ν. ι Μ. ι	14 4	ckness include: a silica layer sive removal.
뷥	822 82	μ. M	Speci	-	~ ~	n -1	ŝ	<u>م</u>	~ 8	a Thi (bMean adhe:

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SPECIMEN DIMENSIONAL CHANGES

TABLE V

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AFML-TR-74-45



Figure 8. Char Depth And Surface Recession For Carbon Phenolics



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Figure 9. Internal 'emperature Histories For R2 Carbon Phenolic



Figure 10. Internal Temperature Histories For R3 Carbon Phenolic



Figure 11. Surface Temperature Histories For Carbon Phenolics

for the R2 specimen. For the measured data a constant grey body condition was assumed and the results were reported as nominally true values. This was considered to be a reasonable first approximation in that a $\pm 25\%$ uncertainty for a representative emittance of 0.8 resulted in a temperature uncertainty of only about $\pm 6\%$.

Although achieved in later efforts, the nominal environmental parameters were not met in the early work reported herein due to equipment and instrumentation limitations (Tables I and VI). In general, the heat flux and heat load for the second step as well as the enthalpy was low. Further, the exposure times were longer and more brief than nominal for the first and second steps, respectively. A higher plenum pressure was used experimentally and sub-atmospheric flow expansion was found in several cases.

The heat flux was lower for R2 as compared to R3 and the enthalpy was lower for R3 relative to R2 (Table VI). The highest plenum pressure, plenum extension pressure, and nozzle pressure at the three stations was generally found for the R3 run. There was no consistent dependency of charring, recession, in-depth response, or surface temperature on the differences in these environmental parameters for the two specimens.

For silica phenolic (Code R1), the approximate mean char depth and corresponding mean recession was 0.200 and 0.077 inches, respectively (Table V). Figure 12 illustrates the profiles for the char, surface recession, and a residual layer resulting from melting of the silica cloth. The mean layer thickness was about 0.014 inches. There was considerable and uneven charring near the leading edge. The recession peaked in the central region of the specimen. The nominal environmental parameters were not readily obtained experimentally (Table I and VI).

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EXPERIMENTAL RESULTS - TWO STEP EXPOSURES

Ctem/Code Composite Type	R1 S/P	R2 C/P	д С/Р
Environment			
Enthalpy, Btu/lb -eat Flux, Btu/sq ft-sec	4540 21.4/276	4900 23.0/255	14186 21.0/312
Pressure, atm Plenum	6.7	5.6	6.8 2.05/1. 2
Extension Mozzlo St 1	1.0/4.2	1.0/4.0	1.0/1.2
	1.0/2.0	0.96/0.75	0.55/1.1
4	1.0/	1.0/0.61	0.95/0.95
Time, sec Total Heat Load, KBtu/sq ft	243/48 5.20/13.2	5.56/13.0	5.10/16.2
Sesponse			
Initial Thickness inch ^a Recession, inch ⁰ Surface Temperature, ^o R	0.555 0.077 1980/4020	0.574 0.056 2090/4040	0.558 0.108 1970/4070
^a Thickness includes adhesive l	ond and aluminum	substructure.	b _{Mean} recession value.

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Figure 12. Char Depth And Surface Recession For R1 Silica Phenolic

The internal temperature histories for the R1 run revealed unexpected high internal heating (Figure 13). Even considering the differences in actual thermocouple depths, the temperatures exceeded those found at any given time for R2 and R3 (Figures 9 and 10). Figure 13 represents an inexplicable "worst case" in that less severe thermal gradients were found for two incompleted runs for this composite. In addition, less internal heating was observed for another silica phenolic composite (C-100-48/DP 25-10) of comparable composition. The recession and second step surface temperatures were higher for the second material consistent with a ligher heat flux for this step. For the second silica phenolic composite, the internal temperature histories were similar in form to that of the carbon phenolic composite R3. The silica phenolic composite temperatures, however, were about 100°F lower at the first thermocouple depth with a smaller difference being observed at greater depths.

The surface temperatures were nearly identical for the Rl and R3 composites for the first step and the difference did not exceed about 400°F for the second (Figure 14). As for the carbon phenolics, the Rl color temperatures were taken as true values by assuming a grey body condition as a first approximation.

A charring-ablation computer code was used to assess the response of a representative carbon phenolic in a nominal two step environment. As the thermophysical properties were not available for the R2 and R3 composites, the major materials property inputs were taken as being similar to those of Table IV. The selected values had the additional advantage of confirmation in other work. In addition to the properties uncertainty, which was most important with respect to in-depth charring and temperatures, the time-resolved environmental parameters were not available for each run. It was possible, however, to average extensive data to provide mean values of enthalpy, heat flux, and pressure. The uncertainty in these parameters was significant primarily with respect to recession and surface temperature.



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Figure 13. Internal Temperature Histories For R1 Silica Phenolic

AFML-TR-74-45



Figure 14. Surface Temperature History For Carbon Phenolic (R3) And Silica Phenolic (R1)

A calculated recession of 0.11 inches was essentially identical to the R3 mean value but high with respect to the R2 specimen (Figure 8, Table VI). The calculated char depth of 0.365 inches exceeded that found for both carbon-phenolic specimens (Figure 8, Table VI). The calculated recession histories were nearly linear for the two steps; the char histories were less linear reflecting the transient nature of the environment. Both cases were averaged to give mean linear values for step 1/step 2. The results were 0.00010/0.0016 inches/sec and 0.00092/ 0.0026 inch/sec for the recession rate and the charring rate, respectively.

The calculated internal temperatures were in fair agreement with results for the R2 run; there was considerable difference with respect to the R3 specimen (Figures 9, 10, and 15). The good agreement between the calculated and actual R3 recession as compared to the differences in in-depth charring and temperature could not be adequately explained. Two important aspects were an experimentally low heat flux for the R2 second step, which resulted in a low recession, and the irregular recession and charring for R3 (Table VI, Figure 8).

The calculated surface temperature history contained a first step transient that was not observed experimentally. The starting value of about 1000°R increased slowly, reaching the experimental level near 2000°R at about 160 seconds, and continued to increase on up to the step change. The differences between the calculated and experimental temperatures were less than a few hundred degrees for the second step. The calculated and measured temperature discrepancy for the first step was not readily resolvable.

The environmental parameter inputs to the code were a more accurate representation of the experimental case than the nominal values (Tables I and VI). The enthalpy was adjusted to 5500 and 4500 Btu/lb for the two steps. The heat fluxes involved 28 Btu/ft²-sec and incremental levels of 360/320/340 Btu/ft²-sec, respectively. These levels were based upon an average pressure history of 1 atm and increments of 0.82/0.82/0.82/0.88 atm for the first and second steps, respectively. A comparison was made



Figure 15. Predicted Internal Temperature History For Carbon Phenolic

between this and an earlier model with the single difference of a constant second step heat flux of 360 Btu/ft^2 -sec. The refined model predicted less recession, more char, and a higher surface temperature for the second step. In addition, a saddle-like change in the surface temperature was found for the refined model but not for the first one. In terms of an experiment, this type of a break, which was observed for R2 and to a lesser degree for R1, appeared consistent with local pressure change resulting from erratic flow expansion, specimen and shock interactions, specimen ablation, or a combination of these effects.

Excluding the apparent normal ablative behavior resulting in some irregularity and porosity of the surface in conjunction with an uneven char layer, there were several experimental artifacts with the potential of contributing to these physical results. The experimental aspects were related to decreasing recession along the specimen during the first step, shock reflections and specimen interactions, downstream effects by ejected upstream contaminants, and heat losses in the nozzle region. Considering these factors, possible interactions, and variability with composite type, it was considered that the most reliable char and recession results corresponded to an average over the specimen mid-region.

Two R2 runs required termination due to equipment malfunctions. One run was ended during the first step; the second, just after bypass closure. The recession generally decreased along the specimen surface, an observation consistent with a low mass flow losing energy down the surface. Expansion was found at the downstream end of one specimen. The actual recession was apparently dependent upon undefined conditions resulting from the malfunction, which consisted of anode failure with water flooding in one case and failure of the bypass plug to seat properly in the second.

The char and recession variability along the length of the specimen was probably initiated during the first step. Uneven recession in the mid-region during the second step was consistent with but not necessarily

totally due to flow environmental changes. A decline in heat flux and enthalpy beyond the mid-region was the expected result of flow expansion in the rectangular nozzle.

The nozzle design condition was for expansion to atmospheric pressure at the exit to provide a parallel flow with minimum shock reflections and specimen interactions. Experimentally, specimen ablation frequently opened the channel and resulted in an exit pressure below atmospheric. The local ablative behavior changed with any significant reflections and interactions. A self-aggravating effect was possible. A technique to reduce underexpansion, an upstream pressure just above atmospheric, was not always feasible due to power control limitations and specimen ablation.

The ablative species generated and ejected into the flow in the upstream region contaminated the downstream flow. The seriousness of the contamination could not be readily resolved. For many specimens, there was some evidence of a heat lcss effect in the nozzle region, a result of excessive conduction into the nozzle hardware.

A run was made for a specimen of graphite (ATJ-S). This material was expected to result in minimum flow perturbation due to the small total recession, to be relatively free of any shock interaction-induced local recession, and to give minimum downstream contamination. The initial thickness range was 0.581-0.614 inches, the recession range was 0.017-0.045 inches, and the peak recession was found at a station 2 inches from the leading edge. There was an increase in total recession from the leading edge to the mid-region and then a decline from this area to the end of the specimen. The graphite results confirmed that irregular recession formed during the first step and that flow expansion effects during the second were viable mechanisms for the composites. Any accelerated recession resulting from shock interactions was considered strongly composite-type dependent. Flow contamination and nozzle heat loss effects were considered secondary to these other mechanisms. Therefore, it was concluded that the most error-free data should be taken as an average for the mid-region of the specimen.

The original air arc heater was modified with auxiliary hardware for this study. This effort required particular invelopmental emphasis for the bypass nozzle, the graphite plug for this nozzle, and the graphite specimen insulator. ŧ

2. MULTIPLE STEP EXPOSURES

R6 carbon phenolic specimens were exposed to multiples of a five step sequence of increasing and decreasing environmental parameters (Table VII). The objective of using sequential steps was to isolate early and late mechanisms and to measure specimen physical change and weight loss.

The mean recession rates, as measured by a photographic technique during the run, were not always consistent with the rise and fall of enthalpy and heat flux (Table /II). For the first and fifth steps, the film image was not bright enough for recession evaluation. The total recession and the total weight loss, as measured for a core taken at the reference station 1.5 inches from the leading edge, were generally consistent with the step changes in environmental parameters (Table V). This conclusion excluded the data for specimens 2, 6, and 7 due to malfunctions for these runs. The differences found between the areal weight change for the core and the total specimen were associated with irregular specimen charring and recession (Table V). The magnitude of the difference tended to follow the increasing and decreasing environmental parameters.

The internal temperature history data were scattered but not unreasonable (Figure 16). The general trend was for steadily increasing temperature with time, with some temporal changes in the rate, at the six thermocouple depths. Consistent with the largest increase in flow energy, the first step change resulted in the largest internal temperature perturbation.

EXI	PERIMENTAL R	ESULTS -	R6 C/P M	ULTIPLE :	Step exf	OSURES			
Item/Specimen ‼umber	step Number	-	2 ⁴	3р	÷	Ś	ee e	pć.	Ø
Enthaloy, Btu/lb	- 9 04 5	1080	1010 3520 2340	1050 3520 5920	1150 3690 6190	1120 3550 5920 3630	4060 1900 1730	5860 3890 1790	1050 3520 5920 7560 1730
Hee', Flux, Btu/sq ft-sec	- 8 6 7 5 5	229	213 473 263 ^a	220 472 656	241 490 689	234 472 654 508	452 253 286	651 517 297	220 457 512 290 290
Sec.	- N 0 4 N	19.78	20.20 14.65 1.16 ³	20.41 15.00 8.230	20.03 14.73 12.89	20.05 14.95 13.08 16.20	13.25 16.05 10.23	13.08 16.43 7.35d	20.00 15.00 13.10 21.03 21.05
Recession Rate4 (inch/sec) x 10-4	トマラキシ	ł	2.70	2.81	3.10	3.25	2.25°	2.93	3.57
Surface Radiation, Rtu/sq ft-sec	- 0 m t v	6	14 197 148a	39 179 313	59 209 465	38 189 403 312	174° 105° 111	300 264 1 <i>5</i> 4	35 193 258 26 26 26
Surface Terperature, ^O ?	ータクキシ	2460	2490 48990 48633	3330 4350 5650	3640 4930 6250	3240 4920 6060 5500	4930° 4300° 4370	5500 5250 4:500	3230 4920 5430 5430
^e Step 3: torminated early virture values. ^e Step 3: ⁶ ters 3, 4: ap 2rc failed ¹ ters 1, 4: ar virturs ir	ranually: h terninated : hert flux	eat flux early nar and spec	probably unlly. incn 1 t	erronso Spectron a probab	us. Tr data fo ly erroi ro ece	ansient or re-e rrous.	surface xfosure to Step 5: rito dota	radlatio. o steps 2 terring.	1 and terp- 2 and 3. 1ted early

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TABLE VII

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the more distinctive response features for the five steps included the following:

Step 1. At expanded and thin char beneath a roughened surface with lic.'s actual recession. The cloth edges were distinctly exposed. The surf is radiation and temperature was not reproducible from run to run. The inregular behavior was tentatively attributed to surface kinetics control of char oxidation and the known kinetic dependences upon surface characteristics in this regime.

Step 2. A thicker char with the onset of recession at a rate of about 0.003 inch/sec. The surface temperature was relatively reproducible for the different runs.

Step 3. The recession rate was essentially the same as for step 2 for his peak heating condition. The exact reasons were not clear. This apparently real recession could have been associated with a constant existive mechanism or an error is the measurement of a transient rate.

Citep 4 An unexpectedly hims eccession for the heating rate. The high-speed surface movies showed an onset appearance of fine-scale missionechanical removal of particles of apparently weakened char.

Scep 5. The estimated recession rate was 0.006 inch/sec, a value higher than for step 4. There was film evidence of increased particle loss in a milder environment. The surface temperature was low, a result consistent with a micromechanical removal mechanism.

Staps 3, 4 and 5. This sequence was run to establish if micromechanical removal effects were inherent for the composite and/or dependent upon prolonged heating uncer steps 1 and 2. Neither the step simulation nor the experimental data were satisfactory. For specimen 7, the recession rate was lower than for specimens with pre-heating under steps 1 and 2. The movies, nowever, revealed a low rate of micromechanical removal, a result consistent with the lower than expected surface temperatures for this rem.

All Steps. The reduction of convective heat flux resulting from surface radiation was approximately 20%, 40%, 65%, 60%, and 30% for steps 1 through 5, respectively.

The response of the R6 composite to the five step sequence was predicted by means of a charring-ablator computer code. The principal thermophysical properties were obtained for a R6 specimen by a nonlinear regression analysis of data for a test in a steady-state environment (Table IV). The nominal values of the environmental parameters were used as inputs ($\dot{T}able$ I).

The calculated recession showed two regions of disagreement with respect to the experimental results (Figure 17). A slightly righ recession predicted for steps 1, 2, and 3 was tentatively attributed to use of a diffusion-control limit on recession in the simplified computer code rather than modelling surface kinetics control. A high recession observed for step 4 and as extrapolated to step 5 was an apparent result of micromechanical surface removal.

The calculated in-depth temperature histories were generally consistent with the experimental results with respect to magnitude and relative shape (Figure 16). There was generally an increase in the rate of temperature rise between 500° and 1200°F. This effect was consistent with and possibly due to endothermic resin pyrolysis and a reduction in local thermal conductivity. The gradual decrease in the rate at higher temperatures was consistent with a possible increase in char thermal conductivity. The experimental rates increased more rapidly at later times than the calculated values, a potential result of micromechanical surface removal.

The predicted surface temperatures were low for steps 1 and 2. This was partially due to improper representation of oxidative reactions for the computer model. The agreement between calculated and experimental values was reasonable for steps 2 and 3. The low experimental results for step 5 were associated with the micromechanical removal mechanism.



Figure 17.

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The multiple step specimens underwent irregular charring and recession with the relative severity increasing with the number of steps. The more reliable ablative information corresponded to the specimen mid-region. The physical measurements were made at the reference station in this area. Based upon the analysis of two step exposures, it was considered that the most important sources of irregular recession involved initiation during early heating and flow expansion effects with some contribution by nozzle measurements.

SECTION VI

SUMMARY AND CONCLUSIONS

An investigation was made of the fundamental response of carbon phenolic and silica phenolic composites under extended heating conditions. Specimens were characterized under stepwise heating pulses of either five minutes (two steps) or up to 1.4 minutes (to five steps) in duration.

The response under two step heating was dependent upon composite type. Considering the influence of artifacts associated with environmental/specimen interactions, the internal and surface temperatures, recession rates, and surface recession patterns were not unreasonably anomalous for one each carbon phenolic and silica phenolic composite.

Tra agreement of charring-ablator predictions for a nominal carbon phenolic in a nominal two step environment with experimental results was dependent upon composite type and the specific test. There was no direct evidence of micromechanical removal of the surface for this low shear environment.

The ablation of carbon phenolic under a five step heating sequence consisted of thermochemical and thermomechanical mechanisms. The char surface was lost by conventional oxidative mechanisms including surface kinetics control at low temperatures. There was micromechanical surface removal under sufficiently severe heating, an effect dependent to some degree upon prolonged early heating. The computer code analysis of response was in fair agreement with experimental results consistent with the code modelling of oxidative mechanisms, the exclusion of micromechanical mechanisms from the code, and the uncertainty in environmental and property inputs.

SECTION VII

RECOMMENDATIONS FOR FUTURE STUDY

The study revealed several deficiencies in securing accurate and useful analytical and experimental results. Specific areas of needed improvement included:

a. Accurate control and reproducibility of environmental variables including the monitoring of the variables during the run.

b. Increased accuracy in the measurement of the environmental variables.

c. Statistical characterization of a material to determine mean response data.

d. Careful apparatus design with a view toward suppressing heat and mass transfer errors. Examples include: continuous specimen repositioning to avoid flow perturbations; minimal flow contamination; minimal shock wave/specimen interactions; optimal channel nozzle design to minimize flow expansion effects; specimen insulation against edge and nozzle heat losses; uniform heat and mass transfer along the specimen.

e. Multiple station monitoring of specimen recession rate and surface temperature.

f. Heat flux and pressure diagnostics for shapes simulating specimen recession patterns.

g. An accurately modelled computer code for response analysis including micromechanical removal modes.

h. Accurate thermophysical properties for the material.