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# CLASSIFICATION OF MATERIALS BY SHOCK PROPERTIES

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Stanford Research Institute
Menio Park, California 94025
Contract F29601-67-C-0087

TECHNICAL REPORT NO. AFWL-TR-69-96

Nevember 1969

AIR FORCE WEAPONS LABORATORY

Air Force Systems Command Kirtland Air Force Base

New Mexico

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AIR FORCE WEAPONS LABORATORY
Air Force Systems Command
Kirtland Air Force Base
New Mexico

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#### CLASSIFICATION OF MATERIALS BY SHOCK PROPURTIES

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#### FOREWORL

This report was prepared by Stanford Research Institute, Menlo Park, California, under Contract #29601-67-C-0087. The research was performed under Program Element 61102H, Project 5710, Subtask RAS1114 (previously 15.018 and 15.025), and was funded by the Defense Atomic Support Agency (DASA).

Inclusive dates of research were May 1967 to No. in 1969. The report was submitted 10 October 1969 by the Air Force Weapons Laboratory Project Officer, Lt Mark Ferdman (WLRX). The former project officer was Captain G. P. Crotwell (WLRP).

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#### ABSTRACT

(Distribution Limitation Statement No. 2)

A new experimental technique was developed for determining loading and unloading stress-volume paths directly from gage measurements; theoretical models were formulated for stress relaxation and the Bauschinger effect: Humaniot information was generated from impact experiments on aluminum alloys, titanium alloys, and a woven quartz-phenolic; and the new experimental technique and Bauschinger calculations were applied to the aluminum alloys. The new experimental technique provides for measurement of complete loading and unloading paths (in t' particle velocity and stress-volume planes) rather than the discrete Hugomiot or release points previously obtained. The technique is applicable to the examination of nonsteady-state, nonisentropic flow, yield point phenomena, strainhardening, the Bauschinger effect and strain-rate (or stress-relaxation) effects. The technique is based on the entire stress or particle-velocity records obtained from a series of gages embedded in a specimen. Four stress relaxation models were implemented in the SRI PUFF wave propagation cole, and computations were made to obtain representative results. Criteria were suggested for selecting a model based on the precursor attenuation and separation of precursor and main wave. The Bauschinger model implemented in the SRI PUFF code exhibits the smooth unloading adiabat and high rarefaction velocity observed in our experiments on 6061-T6 and 2024-T8 aluminum alloys. The marked differences between the wave shapes and attenuation rate from the Bauschinger model and those obtained with the usual elastic-plastic model makes use of the Bauschinger model imperative if unloading and attenuation are of interest. Experiments with T1-50A, T1-6A1-4V, and T1-13Cr-11V-3A1 titanium served to map Hugoniots from 15 to 750 kbar, indicate Hugoniot elastic limits, and show an alpha - omega phase transformation at 50 kbar in Type 50A. Preliminary experiments on a threedimensional woven quartz phenolic resulted in Hugoniot data from 10 to 200 kbar and indicated a shock-wave structure that is very different from that observed in homogeneous solids.

## CONTENTS

	·	
Section	÷	Poor
I	INTRODUCTION	1
	1. Scope	1
	2. Material Behavior Under Study	2
	3. Materials	.1
	4. Background on Stress Relaxation Models	5
	5. Background on Rate-Indapendent Models	11
11	THEORETICAL MODEL DEVELOPMENT	15
	1. Introduction	15
	2. Nature of the SRI PUFF Computer Code	15
	3. Stress Relaxation Models	24
	4. Bauschinger Model	41
111	EXPERIMENTAL METHODS	49
	1. Theory of Data Reduction for Multiple Embedded Gages	49
	2. Experimental Techniques	55
IV	STUDY OF ALUM M	69
	1. Introduction	69
	2. Dynamic Experiments	70
	3. Calculations for 2024-T8 Aluminum	82
	4. Calculations for 6061-T6 Aluminum	88
	5. Summary of Results	92
v	STUDY OF TITANIUM	99
	1. Introduction	99
	2. Dynamic Experiments	101
	3. Initial and Terminal Observations of Shocked Titanium Specimens	110
VI	STUDY OF WOVEN QUARTZ PHENOLIC	115
	1. Introduction	115
	2. Material Description	115
	3. Ges Gun and HE Experiments	117
	4. Theoretical Calculations	125
VII	SUMMARY AND RECOMMENDATIONS	129
	1. Summary	129
	O. Bassemandahtana	2.00

## CONTENTS (Concl.)

Section		Page
APPENDIX I:	TEST RUNS AND LISTING OF THE SRI PUFF 2 COMPUTER	
	CODE	133
APPENDIX II:	QUASI-STATIC AND ACOUSTIC DATA	257
REFERENCES		273
DISTRIBUTION		278

## FIGURES

Figure	<u>P</u> :	age
1	A Distance(X) - Time(t) Diagram Showing Coordinate Layout and Order of Calculations in the SRI PUFF Code	17
2	Common Forms of Pressure and Stress Hugoniots	20
3	Stress Profiles Following Impact in a Material Represented by Model 1 (Simple Anelastic)	31
4	Stress Profiles Following Impact in a Material Represented by Model 4 (Two Parameter)	32
5	Stress Profiles Following Impact in a Material Represented by model 2 (Band)	38
6	Stress Profiles Following Impact in a Material Represented by Model 3 (Gilman)	39
7	Relationship Between Deviatoric Stress and Specific Volume Obtained from 50-kbar Tests in 2024-T8 Aluminum	44
8	Experimental Arrangement for Gas Gun Attenuation Shots	56
9.	Side View of HE Flyer Systems B, D. and E for Generating Uniaxial Strain Loading and Unloading States	59
10	A Distance(X) - Time(t) Diagram Showing Shock and Rarefaction Waves Caused by Impact	60
11	Symmetrical 'mpact Experiment (HE System B) to Determine Hugoniot Data and Shock Wave Profiles	62
12	Messured Unloading Calibration of Manganin Element in Soda-Lime Glass	67
13	Stress-Particle Velocity Hugoniot Data for 2024-T8 and 6061-T6 Aluminum	71
14	Stress-V lune Hugoniot Data for 2024-T8 and 6061-T6 Aluminum	72
15	Typical Manganin Transducer Records Obtained from Impacts in 6061-76 Aluminus	75
16	Interface Stress Histories Recorded by Mangamin-Glass Gages Behind Aluminum Targets (50 Rhar Impacts)	76
17	Interface Stress Histories Recorded by Mangamin-Glass Gages Pahind Aluminum Targets (20 kber Impacts)	77
18	Comperison of Stress Histories Recorded at Aluminum-Glass and Aluminum-Aluminum Interfaces	76
19	Average Instantaneous Impedance p.Dg as a Function of Axial Stress from 50-kber Experiments in Aluminum	78
20	Stress-Particle Velocity Compression and Release Paths	

## FIGURES (Cont.)

Figure		Page
21	Stress-Volume Compression and Release Paths from 50-kbar Experiments in Aluminum	80
22	Stress-Volume Path Reduced from Data for 2024-T8 Aluminum and a Computed Pressure-Volume Relation	84
23	Comparison of Computed and Recorded Stress Histories from Impacts with 2024-T8 Aluminum at 50 kbar	86
<b>24</b>	Comparison of Computed and Recorded Stress Histories from Shot 13,532 in 4.48-mm-Thick 2024-T8 Aluminum Target	89
25	Comparison of Computed and Recorded Stress Histories from Shot 13,532 in 8.96-mm-Thick 2024-T8 Aluminum Target	90
26	Comparison of Computed and Recorded Stress Histories from Shot 13,529 in 25.42-mm-Thick 2024-T8 Aluminum Target	91
27	Stress-Volume Path Reduced from Data on 6061-16 Aluminum and A Computed Pressure-Volume Relation	92
28	Relationship Between Deviatoric Stress and Specific Volume Obtained from 50 kbar Tests in 6061-T6 Aluminum	<b>9</b> 3
29	Comparison of Computed and Recorded Stress Histories From Impacts with 6061-T6 Aluminum at 50 kbar	94
30	Comparison of Computed and Recorded Stress Histories from Shot 13,559 in 8.21-mm-Thick 6061-T6 Aluminum Target	<b>9</b> 5
31	Comparison of Computed and Recorded Stress Histories from Shot 13,559 in 4.86-mm-Thick 6061-T6 Aluminum Target	•
32	Comparison of Computed and Recorded Stress Histories from Shot 13,530 in 25-mm-Thick 6061-T6 Aluminum Target	96 97
33	Stress-Particle Velocity Hugoniot and Release Data for Titanium Alloys	102
34	Stress-Volume Hugoniot Data for Titanium Alloys	103
35	Stress-Particle Velocity Bugoniot Data at Low Stresses for Titenium Alloys	104
36	Stress-Volume Bugoniot Data at Low Stresses for Titanium Alloys	1.05
.37	Manganin Stress Records from Test G589 on Type 50A Titaclum and Test G791 on Ti+6Al-4V Titanium	108
36	Photomicrograph of Type 50A Titonium Symple After Scing Shocked to 56 kber	111

## FIGURES (Concl.)

Figure		Page
39	Photomicrograph of Ti-13V-11Cr-3Al Titanium Specimen After Being Shocked to 70 kbar	113
40	Shock Velocity-Particle Velocity Hugoniot Data for 3-D Weave Quartz Phanolic	119
41	Stress-Particle Velocity Hugoniot Data for 3-D Weave Quartz Phenolic	120
<b>4</b> 2	Recovered 3-D Weave Quartz Phenolic Target and Aluminum Gas Gun Projectile Head Which Impacted It. Impact Velocity = 0.85 mm/µsec, Impact Stress = 37 kbar	122
43	Comparison of Computed and Recorded Stress Histories for Shot 13,422	124
44	Typical manganin wire/C-7 Transducer Records of Stress- Time Profiles Induced in C-7 by High-Stress (HE) and Low- Stress (Gas Gun) Shock in 3-D Weave Quartz Phenolic	126
<b>4</b> 5	Test 1c: Sample Plot of Stress Deviator (Curves plotted by computer from calculated results)	158
46	Test lc: Summarv Plot of All Time Edits	159
47	Test lc: Summary of Stress Histories at Four Jedit Locations	1 <b>6</b> 0
48	Test 2: Summary Plot of Time Edits	168
49	Test 2: Summary of Stress Histories at Three Jedit Locations	169
50	Test 2: Sample Plot of Stress Devistor	170
51	Test 3: Summary Plot of Time Edite	178
52	Tem: 3: Summary of Stress Elstories at Six Jedit Locations	179
53	Test 3: Sample Plot of Stress Deviator	180
54	Test 4: Summary Plot of Time Edits	186
55	Test 4: Summary of Stress Histories at Six Jedit Locations	189
<b>6.0</b>	There do Commin Dict of Common Developer	100

## FIGURES (Concl.)

Figure			Page
57	Test 5:	Summary Plot of Time Edits	198
58	Test 5: Location	Summary of Stress Histories at Six Jedit	199
59	Test 5:	Sample Plot of Stress Deviator	200
60	Test 6:	Summary Plot of Time Edits	208
61	Test 6: Location	Summary of Stress Histories at Six Jeait	209
62	Test 6:	Sample Plot of Stress Deviator	210

## TABLES

Table	<u>e</u>	Page
1	Parameters of Explosively Launched Flyer Plate Systems	58
2	Measured Static Properties of Aluminum Alloys	70
3	Hugoniot Data for 6061-T6 Aluminum	73
4	Hugoniot Data for 1924-T8 Aluminum	74
5	Aluminum Peak Rarefaction Velocities	83
6	Parameters for the Bauschinger Model for 2024-T8 and 6061-T6 Aluminum	87
7	Properties of Tested Titanium Alloys	100
8	Hugoniot Data for Titanium	106
9	Comparison of Static and Dynamic Yield Strength in Titanium	107
10	Rarefaction and Shock Velocities in Titanium Alloys	110
11	Detailed Material Description of 3-D Quartz Phenolic	116
12	"Hugoniot" Data for 3-D Quartz Phenolic	118
13	Aluminum: Number of Specimens Tested	144
14	Titanium: Number of Specimens Tested	145
15	Titanium Data from Static Tests (Average Values)	147
16	Aluminum Data from Static Tests (Average Values)	148
17	Static Test Parameter Variation for Titanium	149
18	Static Test Parameter Variation for Aluminum	150
19	Acoustic Velocity Measurements	153
20	Illustration of Variability of Elastic Constants Derived from Acoustic Velocity Measurements	154
21	Manufacturer's Data for Type 50A Titanium	155
22	Manufacturer's Data for Ti-6Al-4V Titanium	156
93	Manufacturar's Data for Ti-13Cr-11V-341 Titanium	180

## NOMENCIATURE

A'	(24/3)/(E+44/3), an attenuation factor
<b>A</b> .1	(2μ/3) ρ/(K+4μ3) <sup>3</sup>
$A_1, A_2$	functions of $\xi$ in Eq. 34
В	damping coefficient
Bg	fictitious gage velocity associated with propagation of a constant particle velocity level
b	Burger's vector
c	bulk modulus, EQSTC in program; a relaxation constant in Gilman's model
c <sub>o</sub>	coefficient of quadratic artificial viscosity
c <sub>1</sub>	coefficient of linear artificial viscosity
$c_{\mathbf{L}}^{\mathbf{c}}$	acoustic velocity, longitudinal
$^{\mathrm{C}}_{\mathrm{R}}$	rarefaction velocity
cs	acoustic velocity, shear
D	density; coefficient in Eq. 27, EQSTD in program
D <sub>g</sub>	fictitious gage velocity associated with propagation of a constant stress level
E	Young's modulus; internal energy
EH	internal energy on the Hugoniot
Erad	radiated energy
Eref	reference energy
Es	sublimation energy, EQSTE in program
EQSTG	Gruneisen ratio in program
eqstc	bulk modulus ** zero stress
F	(8/34) (dy/dt), the stress relaxation function
g	stress relaxation function; a constant in Band's relaxation model
H	y-1, EQSTH in program
K	bulk modulus
k	constant
k <sub>o</sub>	constant
M	constant in Eq. 4; mass between gages; shear modulus
Mo	initial shear modulus

```
M<sub>1</sub>
           strain-hardening modulus during loading
M<sub>2</sub>
           initial unloading shear modulus
           strain-hardening modulus during unloading
M3
           shear modulus in the computer program
MU
MUIIN
           unloading shear modulus in the computer program
MUo
            initial value of shear modulus
\mathbf{MU}_1 , \mathbf{MU}_2
           coefficients in series expansion of shear modulus
            exponent in equation for shear modulus during unloading
N<sub>1</sub>, N<sub>2</sub>
           number of dislocations; C/(TE_{\alpha}\rho_{\alpha}), EQSTN in program
N
            initial number of dislocations
N<sub>m</sub>
            number of mobile dislocations
           initial number of mobile dislocations
N
           total number of dislocations
N<sub>+</sub>
           initial total dislocations
            value of N at t = t_O + \Delta t
            value of N_t at t = t_0 + \Delta t
            constant in Eq. 8
           pressure
PH
            Hugoniot pressure
            reference pressure
            resultant or total stress
8
            coefficient in Eq. 27, EQSTS in program
           deviatoric stress at end of time increment
SD,
            deviatoric stress in ith direction
SD,
            deviatoric stress at beginning of time increment
SD
            principal stress in the ith direction
s
ii
            relaxation times
            yield relaxation time constant
            relaxation time
Trlx
            time
            time at beginning of calculation cycle
            arrival times of a stress level at positions \mathbf{X}_1 and \mathbf{X}_2
            arrival time of the shock wave
```

```
U
          particle velocity
Ug
          shock velocity
          particle velocity
u
u<sub>o</sub>
          initial particle velocity
          specific volume
          initial specific volume
          specific volume on the Hugoniot
          dislocation velocity
          maximum dislocation velocity
          slope in x,t plane of line of constant particle velocity
X
          position
XP0
          exponent, N_1, in the computer program
XР
           exponent, No, in the computer program
          gage locations
x<sub>1</sub>,x<sub>2</sub>
           yield strength
Y
           initial yield strength
           yield strength at yielding during loading and unloading
Y,Y
           respectively
           average of Y and Y olo
Yavg
Y
           final value of Y at end of time increment
Yold
           value of Y at beginning of time increment
YADD
           strain-hardening modulus in the program
           ratio of strain-hardening moduli during unloading and
YADF
           loading in the program
YO
           Y in the program
           mass of a finite difference cell
Z
           constant in Eq. 6
α
В
          dislocation damping constant
r
           Gruneisen's ratio
           initial Gruneisen's ratio
           ratio of specific heats; plastic shear strain
           strain; a constant in Band's relaxation model
           static strain
```

plastic strain component

```
θ
            SD/|SD|
            Heaviside constant
            Lame's constant, a modulus
            Lame's shear modulus; \rho/\rho_0^{-1}, strain
            strain at maximum stress
            Poisson's ratio
v
            |SD| -2/3Y, excess of deviatoric stress above shear
            strength
ρ
            density
            initial density
            density at yield during loading
ρ<sub>1</sub>
            density at unloading
\rho_2
            density at maximum stress
\rho_{\mathbf{m}}
σ
            stress
            constant in Eq. 8; HEL after stress relaxation
\sigma_{o}
\sigma_{\tt Hel}
            Hugoniot elastic limit
           Hugoniot elastic limit as a function of time
\sigma_{\mathbf{i}}
            initial amplitude of precursor
\sigma_{\alpha}
            3/4 SD, shear stress
7
           Y/2, shear stress to cause yielding
	au_{
m b}
To
           constant in Eq. 5
           constant in Eq. 4; work-hardening coefficient in
           Gilman's model
```

AFWL-TR-69-96

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#### SECTION I

#### INTRODUCTION

#### 1. Scope

The gross features of shock wave propagation in solids appear to be fairly well understood. These features generally include an elastic precursor, a main (or plastic) shock wave, and a rarefaction or unloading wave. The simplest and most common model incorporating these effects is the model of ideal plasticity. For precise quantitative prediction of these waves, however, theoretical calculations based on the model of ideal plasticity are inadequate. The behavior of real materials is modified by stress-relaxation (time-dependent) effects, Bauschinger effect (difference in load and unload response), a nonlinear transition from elastic to plastic behavior, strain or work hardening, etc. (These terms are explained more fully later in this chapter.) Each of these effects is important for some materials. Each leaves its imprint on some portion of the wave profile and leads to a modification of the attenuation of a short-duration shock pulse.

The main objective of the present program was to develop methods for characterizing materials by their shock propagation behavior. These characterizations fall into two major categories:

- . Rate dependent: stress relaxation effect.
- . Rate independent: Bauschinger effect on unloading and a nonlinear transition from linear elastic to plastic behavior on loading.

#### Specific objectives were to:

- . Study the nature of stress relaxation effects in swock waves.
- . Investigate unloading or Bauschinger effects.
- . Relate shock properties to quasi-static and acoustic material properties.
- . Improve methods to experimentally study rate-dependent and rateindependent effects.
- . Develop or improve theoretical models for both rate-dependent and rate-independent effects and implement them in a wave propagation computer code.

Develop guidelines for ascertaining the relative importance of the several effects studied.

These objectives were pursued in an experimental study of three materials: aluminum (2 alloys), titanium (3 alloys), and a woven quartz phenolic. The theoretical study, which was directed mainly to the aluminum, led to the exploration of five computational models, which were incorporated into the SRI PUFF computer code.

The two main developments of this program are the theoretical models implemented in the SRI PUFF code and the development of a new method for obtaining complete loading and unloading paths from experimentally recorded stress-time profiles. These two topics occupy the Sections II and III of this report. Sections IV. V, and VI contain descriptions of each material studied. In the materials sections the experiments are described, the data analyzed, and comparisons are made with the theoretical models.

#### 2. Material Behavior Under Study

The kinds of behavior studied in this program fall into the two broad categories: rate-dependent and rate-independent. The rate-dependent phenomena are associated with an apparent softening of a loaded material with time. If the material is loaded with a stress that remains at a constant level for some time, there is an instantaneous strain followed by a gredually increasing strain as long as the load remains: this response is called creep. Alternatively, one may observe that the instantaneous stress required to produce a constant strain in the material gradually decreases with time: this response is called stress relaxation. When the material is loaded at different strain rates, the successive states describe different stress-strain paths: this behavior is referred to as strain rate effects. Stress relaxation, creep, and strain rate effects are just different (but relatively interchangeable) descriptions of the basic time-dependent phenomena.

<sup>\*</sup> The term "creep" is usually associated with very long time intervals and slow loading rates. However, the concept of time-dependent increasing strain at constant stress is also applicable to shock loading rates. In fact, some analytical creep functions (such as Gilman (Ref. 5)) have been successfully used to predict time-dependent phenomena in shocks.

These time-dependent phenomena lead to predictable variations in the shape of shock waves. A common effect is the attenuation of the amplitude of a flat-topped wave or a precursor with distance. Another effect is the variation of wave front shape with propagation distance, particularly a broadening of the shock front and general rounding of the stress-time profiles. While the presence of this latter effect may suggest time-dependence of the material properties, this effect is not conclusive evidence of time-dependent material response. Time-independent material properties can also cause such an effect to occur.

Time-independent phenomena of particular interest are work hardening, the elastic-plastic transition, and the Bauschinger effect. The term work hardening (or strain hardening) refers to the tendency of the yield strength of a material to increase as the material is strained. The hardening is often characterized by a work-hardening modulus relating the yield stress to the strain. The elastic-plastic transition refers to the region of the stress-strain curve where the material behavior is not linearly elastic nor purely plastic but is a mixture of nonlinear elastic and plastic. In a material such as aluminum in which no sharp yield point occurs the elastic-plastic transition describes the "gradual yielding" observed. This transition region is convex upward on the stressvolume plane and therefore leads to dispersion of compressional waves, an apparent decrease in the Hugoniot elastic limit (HRL) with distance, and a nonsteady transition degion between a precursor and a main wave. (These are some of the same effects usually attributed to time-dependent material properties.)

The Bauschinger effect refers to a change in the stress-strain relations that occurs when the sign of the stress is reversed after partial loading. Usually this effect is interpreted as simply a rejection in yield strength, although bulk and shear moduli may also be affected. The present study of shock wave propagation has indicated that in fact, the entire shapes of the stress-strain relations may chang when unloading occurs after shock compression. Because the Bauschinger effect comes into play only on unloading in the present case, it does not modify wave fronts. Instead the effect modifies the arrival time of rarefaction waves, the attenuation rate of stress waves, and the shape of the "back,"

<sup>\*</sup> The REL is the maximum axial stress that the material can support (in uniaxial strain and under shock loading conditions) with a limearly elastic response. Such a limit of elastic behavior under static test conditions is referred to as the "proportional limit."

or unloading portion of the waves.

Hugoniot data may be considered part of the time-independent phenomena of interest to us on this project. A Hugoniot point is the final equilibrium state (hence, time-independent) of a material that has been traversed by a steady-state shock wave. For convenience, we have adopted the common practice of generalizing the definition to include any approximately steady-state stress wave, such as the gradually rising stress waves seen in porous materials, composites and rate-dependent media.

Real materials exhibit complex behavior that is a mixture of timedependent and time-independent phenomena. Hence, some materials may show a combination of all the above effects.

#### 3. Materials

Three materials were used in this study as the basis for theoretical model development and investigation of shock properties. The primary material was aluminum, with preliminary studies being performed also on titanium and a woven quartz phenolic.

Two alloys of aluminum-6061-T6 and 2024-T8-were selected especially to study stress relaxation phenomena. On the basis of earlier work, it was expected that the 2024-T8 would show significant stress relaxation and that 6061-T6 would not. In addition, we planned to investigate in detail the unloading behavior of these alloys because of its relevance to experimental studies of other materials. Considerable information on the loading behavior of these alloys was available so that we were able to deal immediately with unloading phenomena and precursor decay.

The preliminary study of titanium was undertaken to map the lowpressure (under 500 kbar) Hugoniot, to look for possible high-pressure
phase transitions, to investigate strain-rate sensitivity, and to study
release waves and attenuation. Three titanium alloys were selected: Ti50A, commercially pure titanium all in the alpha phase; Ti-6Al-4V, an alloy
with mixed alpha and beta phases; and Ti-13Cr-11V-3Al, an alloy that is
entirely beta phase. The Ti-6Al-4V alloy was selected because it is a
material of practical importance that has already been studied at
intermediate strain-rates, where it exhibited strain-rate sensicivity.
The Ti-50A and Ti-13Cr-11V-3Al alloys were chosen to study independently
the behavior of each of the constituent phases of the Ti-6Al-4V alloy.

The quartz phenolic was supplied to SRI by the Air Force Weapons Laboratory for a preliminary investigation. Although the effort with this material was much more limited, the purpose was essentially the same: to perform a series of experiments and attempt to establish which shock phenomena are important.

#### 4. Background on Stress Relaxation Models

Early work on stress-relaxation was presented by Malvern in 1951 (Ref. 1). He derived basic relaxation equations that are essentially the same as those corrently used. His formulation, which is for a rod with stress-free sides, is

$$\frac{\partial \sigma}{\partial t} = \frac{E \partial \epsilon}{\partial t} - g(\sigma, \epsilon) \tag{1}$$

where

o,f = stress and strain in the direction of the shock motion

E = Young's modulus

g = the relaxation function

t = time

Malvern solved a simple impact problem using

$$g = k \left[ \sigma - f(\epsilon) \right]$$
 (2)

where k is a constant. This model is one form of the standard linear anelastic model described by Zener in 1948 (Ref. 2). In Malvern's model the relaxation time is proportional to 1/k, a constant, and is there're independent of stress lev 1.

In 1963, Taylor (Ref. 3) performed a series of impact experiments on Armoo iron. The results showed that relaxation was very rapid at high stresses and slower at low stresses. The variation of precursor amplitude with distance of travel was independent of the initial impact stress except for distances less than 2 mm. These experimental data did not correlate well with predictions of the standard model that uses a constant relaxation rate independent of initial stress level. Taylor constructed a stress-relaxing model by making k in Eq. 2 a function of the square of the impact velocity. The predictions of this model were in reasonable agreement with his data.

In 1964 Lublimer (Ref. 4), presented a general discussion of methods for solving wave propagation problems in rate-dependent and rate-independent

materials. He noted that characteristic lines are usually curved for rate-dependent materials, and for both types of materials there is a discontinuity in slope between loading and unloading characteristics.

In 1965 a basic paper on the formulation of a dislocation model for stress relaxation was presented by Gilman (Ref. 5). The model of the stress relaxation term is presented as  $\dot{y}$ , the plastic shear strain rate. He introduced

where

N = the number of dislocations per square centimeter

b = the Burger's vector

v = the dislocation velocity

and then derived equations for N and v from the available experimental data. These equations are

$$N = (N_o + My, e^{-\beta y})$$
 (4)

$$v = v_0 e^{-\tau} o^{/\tau}$$
 (5)

where

 $N_{\Delta}$  = the number of initial dislocations

 $v_0 =$ the maximum velocity

7 = the shear stress

M,  $\emptyset$  and  $\tau_o = constants$ 

The model was intended to be applicable to both creep and shock front behavior. Computations showed that, according to the model, there is an incubation time during which dislocations multiply but no significant yielding occurs. Subsequently, the yielding occurs rapidly for a time and then reaches a third, or equilibrium, state when yielding ceases.

Taylor (Ref. 6) reanalyzed his data of 1963 two years later with the aid of Gilman's model above. He sugmented the model t include work hardening, using

$$\dot{\gamma} = \text{LNV} = \text{b(N}_0 + \alpha \gamma) \text{ v}_0 e^{-(\tau_0 + \beta \gamma)/\tau}$$
 (6)

where

y = the plastic shear strain

 $\alpha, \tau_{\alpha} = constants$ 

Ø = a work-hardening coefficient

The constitutive equation for the material was written in the form

$$\dot{\sigma} = (\lambda + 2\mu) \dot{\epsilon} - 8/3\mu \dot{\gamma}$$
 (7)

where

 $\lambda, \mu$  = Lame's constants

 $\sigma, \epsilon$  = stress and strain

The first term on the right-hand side represents elastic strain; the second is the plastic strain. With the use of Eq. 6 for  $\dot{\gamma}$ , Eq. , became a stress-relaxing constitutive relation. With this equation for  $\dot{\sigma}$ , the relaxation time decreases exponentially with stress and therefore has the appropriate form to fit Taylor's data on Armco iron. Taylor considered only the rate of precursor decay, not the changes in the entire wave form.

In 1965, Dorn, Mitchell, and Hansen (Ref. 7) presented a summary of current information on athermal, thermally activated, viscous, and relativistic effects associated with dislocation mechanisms. For each of the first three effects they suggested several possible micromechanisms. It seems likely that at relatively low strain rates (<100/sec) thermally activated mechanisms govern dislocation velocity. But at higher strain rates the velocity is proportional to the excess of shear stress above the static limit: this proportionality suggests a viscous damping mechanism. While it was theoretically predicted that dislocation velocities could approach but not exceed the shear wave velocity, there was no experimental evidence to support this view. No dislocation velocities had yet been measured above half the shear velocity.

Maiden and Green in 1966 (Ref. 8) reported results of an experimental study of rate sensitivity in 6061-T6 and 7075-T6 aluminum, annealed Ti-6A1-4V titanium, and several other materials. The testing rates ranged from 10<sup>-3</sup> to 10<sup>4</sup>in./in./sec. A comparison of stress-strain curves derived from tests up to 10<sup>3</sup> in./in./sec showed no rate sensitivity in either aluminum alloy but considerable rate sensitivity in titanium. In the titanium, the strain rate effect was described by equations of two forms:

$$\sigma - \sigma_{\mathbf{g}} = k_{\mathbf{o}} \log \left( \hat{\epsilon} / \hat{\epsilon}_{\mathbf{g}} \right) \tag{8}$$

$$\sigma - \sigma_s = k_s (\epsilon)^n \tag{9}$$

where

or = stress at a given strain

 $\sigma_{e}$  = the static stress at the same strain

 $\dot{\epsilon}$  = the strain rate

¿ = the "static" strain rate

n,k = constants

The data fit the second form best, with n = 1/5. The corresponding forms of g (see Eq. 1) are

$$g = k_a e^{(G - \sigma_S)/\sigma_O}$$
 (8a)

$$g = k_3 (\sigma - \sigma_8)^5$$
 (9a)

where the k's and G are constants. Both of these forms indicate a relaxation time which is very short for high stresses, longer for low stresses. Equation (8a) is reminiscent of Gilman's model, (Eq. 5), while Eq. 9a exhibits a time constant inversely proportional to stress to the fourth power.

In 1966, Ahrens and Duvall (Ref. 9) reported some stress relaxation studies in Sioux quartzite. They derived a complex form for g from their data. The precursor appeared to attenuate linearly with distance in the rather narrow range for distances which they used.

Butcher and Munson in 1967 (Ref. 10) conducted tests on 1060 aluminum and 4340 steel, both of which are known to be rate-sensitive. Results showed precursor decay and separation of the precursor and main wave. In several stress-time records there was a dip in stress between the two waves. The attenuation appeared to be approximately exponential with distance. Approximate theoretical computations of precursor decay were made using several possible relaxation models. Best results were obtained using Gilman's model with plastic shear strain rate in the form

$$\dot{\sigma} = bNv = bN (1 + \alpha\gamma)e^{-\beta\gamma}v e^{-7}o^{/7}$$
 (10)

In 1966 Johnson (Ref. 11) investigated three models: the standard anelastic model, Gilman's model, and a model suggested by Band (Ref. 12). Johnson provided a derivation of each model and modified them appropriately for application to a material with a specified static yield strength. His basic equation, corresponding to Eq. 7, for one-dimensional flow is

$$\frac{\partial y'}{\partial t} = 4/3 \frac{\mu}{\rho} \frac{\partial \rho}{\partial t} - 8/3 \mu \frac{\partial \gamma}{\partial t}$$
 (11)

where

 $\sigma' = \sigma - p$ , the deviatoric stress

p = the density

u = the shear modulus

y = the plastic shear strain

The form of Eq. 11 emphasizes the fact that only the deviatoric stress is being relaxed; the pressure component follows an elastic, rate-independent, law. Johnson showed that for many forms of  $\gamma$ , the precursor decay could be found simply from

$$\frac{d\sigma}{dt} = -4/3 \mu \frac{\partial \gamma}{\partial t}$$
 (12)

where

σ = the stress at the precursor

t = the arrival time of the precursor

y = plastic shear strain evaluated from the unshocked state

In the modified Gilman model presented by Johnson, dislocation velocity is

$$v = v_0 \exp \left(-\frac{\tau_0}{(7 - Y/2)}\right)$$
 (13)

where Y = the static yield strength. Band's model, as presented by Johnson, has a form identical to the modified one of Gilman, except that the number of mobile dislocations is given by a set of simultaneous equations. These equations express rates of growth, pinning, and annihilation of mobile and total dislocations.

In 1967 Anderson et al., (Ref. 13) conducted impact experiments on 2024 aluminum and polyethylene and measured precursor attenuation in the aluminum. The experimental results were compared with predictions based on four models: the three treated by Johnson plus a new two-parameter model. The new model was based on the constitutive relations

$$\frac{\partial \sigma}{\partial t} = K(\rho) \frac{\partial \rho}{\partial t} - \frac{\sigma - \sigma_1}{T}$$
 (14)

$$\frac{\partial v_1}{\partial \dot{v}} = -\frac{\sigma_1 - \sigma_0}{T} \tag{15}$$

where

T, T = relaxation times

 $K(\rho)$  = the elastic modulus function

 $\sigma_i$  = the Hugoniot elastic limit, a function of time

= the Hugoniot elastic limit after complete relaxation.

This model was formulated to provide both for the phenomenon of precursor decay and for separation of the two waves.

In 1967, Wilkins (Ref. 14) described several possible stress-relaxation functions. He presented typical results from one- and two-dimensional wave propagation calculations.

Historically, the first matter of interest was the rate of precursor decay. This could be measured quantitatively and was clearly an indication of strain rate effects. Stress-relaxation models from various other applications were brought in to explain the decay. As more information became available from dislocation dynamics, more appropriate models were developed to predict precursor decay. Interest was then kindled in describing the entire wave front: precursor, main wave, and region between these waves. At present, models are just being developed to represent all these phenomena.

### 5. Background on Rate-Independent Models

Attenuation of shock waves occurs when the unloading or rarefaction portion of the wave overtakes the shock front. Therefore, the relative magnitudes of the shock velocity and rarefaction velocity govern the attenuation. Early studies of attenuation considered only a hydrodynamic model. In later studies, which are discussed below, it was presumed that the rarefaction behavior is a function of both elastic and plastic effects.

The groundwork for considering elastic-plastic behavior of materials was laid by Morland (Ref. 15) with his analysis of wave propagation in rods. He used the ideal plastic model for his work.

In 1960, Al'tshuler et al., (Ref. 16) used plate slap experiments to study rarefaction velocities up to pressures of 3 Mbar. They employed the stress attenuation rate as a means of measuring the rarefaction velocity. Even at the highest stresses, they noted the presence of initial elastic rarefaction waves. However, they disregarded these waves and concentrated on the slower-moving plastic rarefactions.

Curran (in 1963, Ref. 17) conducted a series of plate slap experiments at lower stress than Al'tshuler et al. used to study specifically the elestic rarefaction waves. His results showed clearly that el stic-plastic effects led to much higher attenuation rates than predicted by hydrodynamic models. He was able to match 'e attenuation rate in computations by assuming a large (tenfold) increase in yield strength during compression.

Barker, Lundergan, and Herrmann (Ref. 18) used plate slap experiments up to 20 kbar to determine the loading and unloading stress-strain relation. They noted a small strain rate effect during loading and a significant Bauschinger effect during unloading. Even at the low stress of 10 kbar, the elastic rarefaction velocity was some 5 percent higher than the initial loading velocity. Under static loading, the Bauschinger effect in aluminum had been well documented by Buckley and Entwistle in 1956 (Ref. 19). In the static work there was an elastic unloading to zero stress. Reloading in the opposite sense followed a stress-strain relation with a much reduced slope. The shapes of these static reloading waves appeared to be geometrically similar for different yield strength levels.

Jones and Holland (in 1964) conducted plate slap experiments with SAE 1018 steel in both the annealed condition and with 20 percent cold work (Ref. 20). The precursor (and hence the yield strength) was about half as large in the cold-worked material as in the annealed: this was interpreted as evidence of a strong Bauschinger effect.

Further evidence that static stress-strain data are pertinent to dynamic studies was gained by Butcher and Canon (Ref. 21) in 1964. They performed plate impact tests on 4340 steel to determine the loading stress-strain relation. There were important effects of work hardening and some stress-relaxation effects, but the authors concluded that the static and dynamic constitutive relations could be taken as identical without serious error.

In 1967, Erkman and Christensen (Ref. 22) conducted plate slap experiments on aluminum at 110 and 340 kbar to study unloading velocities and attenuation. Calculations were made with the ideally plastic model and with an plastoplastic model in which shear modulus and yield strength varied with density. The best correlation with experimental attenuation rates was obtained by sysuming a yield strength that increased threefold. A sharp step in the unloading wave (which is predicted by the elastoplastic model) was not observed in the experiments.

In 1968, Barker (Ref. 23) presented a discussion of plate slap experiments in which a more refined rear-surface measurement technique was used. The data led him to the formulation of a Bauschinger model in which there was an initial elastic release to zero deviatoric stress. Then for continued unloading, the yield stress was allowed to increase gradually according to the expression

$$Y = 4/3Y_0 (1 - e^{-800\xi}p)$$
 (16)

where

Y = the initial yield strength

\* the plastic strain component

Calculations with this model compared well with experimentally recorded wave shapes.

The bistory of attenuation studies shows a gradual realization of the complexity of rarefaction waves. First, hydrodynamic calculations were

used, found inadequate, and replaced by ideally plastic models. The next step was to construct ad hoc models that would exhibit some type of Bauschinger effect. In this report we present a Bauschinger model in a general form that should be applicable to a wide range of materials.

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#### SECTION II

#### THEORETICAL MODEL DEVELOPMENT

#### 1. Introduction

Four models for stress-relaxation and one model constituting a Bauschinger effect are outlined in this section. All of the models are sufficiently complex that it is necessary to implement them in a wave propagation computer program to study their effects. The models were inserted into the SRI PUFF code, an artificial viscosity code originally developed for the study of porous materials. These radels affect only the computation of deviatoric stress; so the basic program remains unaffected. As background for the introduction of the models, the nature of the code is presented in this section; the models are then described.

#### 2. Nature of the SRI PUFF Computer Code

#### a. Introduction

The SRI PUFF code is a program for analyzing eno-dimensional wave propagation caused either by the deposition of radiation in the materials or by the impact of two materials. The code contains a novel integration scheme for solving the governing equations of motion. The equations of state for solid, porous, liquid, and gaseous materials are provided. In the present project, the available equation of state routines were augmented to include stress-relixation and Bauschinger effects. As background for the discussion of these effects, brief descriptions of the integration procedure and of the equation of wtate for a solid are given.

#### b. Integration Scheme

The solution procedure for the governing equations of motion is called the method of artificial viscosity. With this method no discortinuous shock fronts occur, but a shock is represented by a stress wave front aproad over 3 to 5 cells. Hence, the equations of continuous flow are applicable throughout the flow field.

The continuous flow equations are the ose-dimensional Lagrangian equations of motion. These equations are reduced to a form in which only first-order derivatives occur. To improve accuracy and stability, integral

statements of the equations are used in preference to differential equations wherever possible.

The following set of Lagrangian equations are merely presented; they were derived by Richtmyer (Ref. 24) and many other authors.

$$X_1 = X_0 + 1/2 (U_0 + U_1) \Delta t$$
 (velocity) (17)

$$U_1 = U_0 + \frac{\Delta R}{D \Delta X} \Delta t \qquad (momentum) \qquad (18)$$

$$D_{1} = \frac{Z}{\Delta X + \Delta U \Delta t}$$
 (mass) (19)

$$E_1 = E_0 + R (1/D_0 - 1/D_1) + E_{rad}$$
 (energy) (20)

$$R_1 = F(E_1, D_1, ...)$$
 (equation of state) (21)

where

X = coordinate location

U - coordinate velocity

D = density

E = internal energy

R = total mechanical stress

 $\Delta u = (\partial u/\partial x) \Delta x$ 

t = time

E and internal energy added by radiation

 $z = cell mass (D \Delta X)$ 

Subscripts 0 and 1 refer to times t and  $t_1 = t_1 + \Delta t$ . The nonsubscripted R and D values appearing in the equations are chosen to provide an accurate representation of the conservation laws.

For convenience in visualizing the progress of the computation, and distance—time (X-t) plot with coordinate points, cells between coordinates, and time steps is shown in Fig. 1. The computations proceed from left to right, one coordinate at a time, updating coordinate location, velocity, density, etc., to the new time  $t_1$ . (Order of the computations is given by the numbers in Fig. 1.) The integration scheme

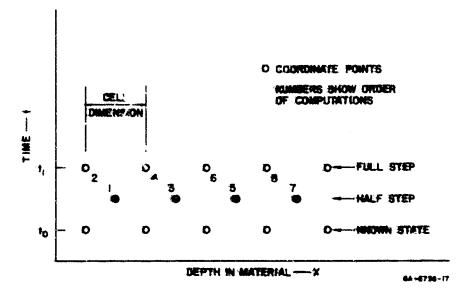


FIGURE 1 A DISTANCE (x)-TIME (t) DIAGRAM SHOWING COORDINATE
LAYOUT AND ORDER OF CALCULATIONS IN THE SRI PUFF CODE

used in the SRI PUFF code is a mixture of the leapfrog method proposed by von Neumann and Richtmyor (Ref. 25) and the two-step Lex-Wandroof method (Ref. 24).

In the leapfrog scheme the basic computation cycle proceeds as follows:

That is, density is computed from the coordinate position, amergy and stress from density, velocity from stress, and coordination location from velocity. The cycle is then repeated in succoeding time steps. The natural procedure, as used in the leaping scheme, is to associate X and U with the coordinate points and to associate the density, energy, and stress with the midcell points (hence, the subscripts c and m in the cycle notation). The midcell quantities are also associated with a time 1/2 \( \Delta \) shead of the X and U values. Thus, the midcell quantities are evaluated at the "alfatep location" in the X-t plot of Fig. 1. In the two-step lax Wendroff method, all quantities are computed at both the coordinate and midcell points. Quantities occurring at coordinates

(and full time steps) are treated as primary; the halfstep (midcell) quantities are secondary and are computed by more approximate equations.

In the SRI PUFF code the basic  $X_c = D_m - R_m - U_c - X_c$  cycle is treated as primary. In addition a subsidiary cycle  $x_m - D_c - R_c - U_m - X_c$  is used to generate the other quantities so that all quantities are available at both coordinate and midcell points. The main advantages of the method are (1) elimination of an iterative solution for stress usually required with the leapfrog scheme thereby decreasing computation time), and (2) diminution of the oscillations in the stress at shock fronts that commonly occur with the Lax-Vendroff method.

#### c. Equation of State

The equations of state considered here relate the stress in a material to its internal energy, density, and the previous state. The material may be in solid, liquid, or gaseous states.  $^*$ 

For convenience in formulating the equations of state for both solid and porous materials, the stress tensor will be defined as the sum of a pressure and a deviator stress tensor. The pressure is defined as

$$P = \frac{1}{3} \left( S_{11} + S_{22} + S_{33} \right) \tag{22}$$

where the S's are stresses on any three mutually orthogonal planes. The deviator stress is the variation of any normal stress from the average:

$$SD_{11} = S_{11} - P \quad (1 = 1, 2, 3)$$
 (23)

For the one-dimensional strain case to be treated here, the stresses can be conveniently taken in the directions of principal stresses, so that only normal stresses occur. Furthermore, the deviatoric stresses are simply related as follows

$$SD_{22} = SD_{33} = -\frac{1}{2} SD_{11}$$
 (24)

<sup>\*</sup> The code also provides for materials that are initially porous. Spalling, separation, and recombination, and thermal strength reduction are also harded by the code. These provisions, which are not of direct interest here, are discussed in Reference 26.

so that the only stress quantities to be computed are SD and P. The pressure is computed as a function of two or more of the other thermodynamic quantities. The deviator stress is computed from a stress-strain relation. The stress is then found as a simple sum of SD and P. With this separation of stress in D two components, the development of an equation of state requires the construction of two relationships, one for pressure and one for deviator stress.

# (1) Pressure

The equation used here for pressure is of the form

$$P = P(E, V) \tag{25}$$

which says that pressure is a function of specific internal energy and specific volume only. The thermodynamic quantities entropy and temperature are not considered explicitly. Equation 25 defines a surface in E-P-V space.

An equation of state represents equilibrium states. Therefore as a material undergoes gradual changes, such as heating, compression, etc., the successive states describe a path on the equation-of-state surface if there is no heat conduction, stress relaxation, or other nonequilibrium process occurring. If the material is compressed by passing through a steady-state shock front and the initial and final states are equilibrium states, then these states lie on the equation-of-state surface. These initial and final states are connected by a straight line, the Rayleigh line, which lies on or above the surface, for the usual, concave-upward, surfaces. The states of transition within a shock front are not states of thermodynamic equilibrium and hence do not necessarily lie on the surface.

Shock experiments lead to the determination of a Hugoniot or Rankine-Hugoniot curve which is a line on the equation-of-state surface. This line is the locus of final states that can be obtained by a steady-state shock transition from a given initial state. A common form for a pressure-volume Hugoniot is shown in Fig. 2.

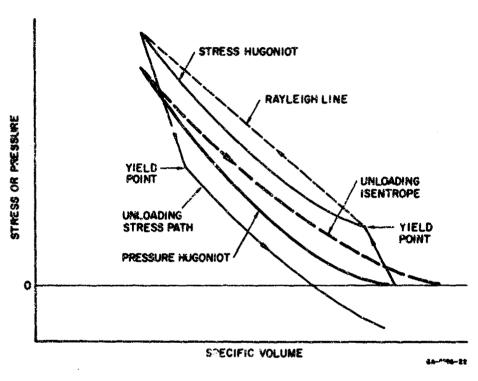


FIGURE 2 COMMON FORMS OF PRESSURE AND STRESS HUGONIOTS

During compression then is some increase in internal energy, so that the Hugoniot does not lie in a single P-V plane.

As a reminder of the role of stress in the compression of the solid, consider the stress-volume Hugoniot of Fig. 2. During compression the stress is greater than the pressure; on unloading, the stress decreases rapidly to yielding and then follows a stress adiabat below the pressure adiabat. The unloading adiabat lies to the right of the Hugoniot for materials that expand during heating. For such materials less internal energy is released by the adiabatic decompression (expansion) than was absorbed during the shock compression.

The pressure equation of state for a solid and the subroutine for calculating it in SRI PUFF are essentially the same as those in PUFF 66 (Ref. 27). The equation of state is described by two analytical equations (one for compression and one for expansion) and is bounded for negative stresses by a spall criterion.

The equation used to describe compression is the Mie-Grüneisen equation

$$P - P_{REF} = \frac{\Gamma(V)}{V} (E - E_{REF})$$
 (25)

where

PREF and EREF = a point on some reference curve at the same specific voltage V

r(v) = the Gruneisen ratio.

Equation 26 has been derived on the assumption that  $\Gamma$  is a function of V only. Equation 26 provides a means for extrapolating the information of a known P-V relation (such as a Hugoniot) to other values of internal energy. Because the Hugoniot is the P-V relation that is most likely to be known, the computations are constructed so that the Hugoniot is the reference curve used. The Hugoniot P-V equation is presumed to be in the form

$$P_{H} = C\mu + D\mu^{2} + S\mu^{3}$$
 (27)

where

$$u = \frac{\dot{\rho}}{\rho_{o}} - 1 = \frac{v_{o}}{v} - 1$$

The internal energy along the Hugoniot is

$$E_{H} = \frac{1}{2} P_{H} (V_{O} - V_{H})$$
 (28)

Equation 28 is based on the assumption that the initial internal energy is zero and that the Hugoniot is concave upward. In general, the latter assumption excludes consideration of phase changes. Also, the relation is strictly true only if the stress Hugoniot coincides with the pressure Hugoniot; however, at high pressures there is usually little inaccuracy introduced by this approximation. With the aid of Eqs. 27 and 28, the Mie-Grüneisen equation takes the following form in the program:

$$P = (C\mu + D\mu^2 + S\mu^3) * \left(1 - \frac{\Gamma\mu}{2}\right) + \Gamma\rho E$$
 (29)

In the computer program the Grüneisen ratio  $\Gamma_{\rm o}$  at initial density is taken as a constant, EQSTG. Then  $\Gamma$  is treated as a function of density such that  $\Gamma_{\rm p}$  is constant.

At a constant volume, Eq. 29 is a linear relation between pressure and energy; hence, constant volume lines on the equation-of-state surface are straight lines. The Mie-Grüneisen equation of state is used for densities greater than the initial density. Thus on the equation-of-state surface the straight line  $V = V_0$  is the boundary between the Mie-Grüneisen equation and an expansion equation.

The expansion equation, which is unchanged from PUFF 66 (Ref. 27), meets four requirements:

- It joins smoothly to the Mie-Grüneisen equation along  $V = V_0$ .
- It expands like PV =  $(\gamma 1)E$  at large expansions (like a gas).
- It provides a linear relation between P and E for constant V.
- It accounts for the partition of internal energy into components for kinetic energy and for intermolecular bond disintegration (sublimation).

An equation that satisfies these requirements is

$$P = \rho \left[ H + (\Gamma - H) \left( \frac{\rho}{\rho_o} \right)^{\frac{1}{2}} \right] * \left\{ E - E_S \left[ 1 - \exp \left( N \left( 1 - \rho_o / \rho \right) \rho_o / \rho \right) \right] \right\}$$
(30)

where

p = density

 $\rho_{0}$  = initial density

I = Gruneisen ratio

H =  $\gamma - 1 = \frac{C}{D} \frac{C}{V} - 1$  for expansion at low densities

 $N = C/(\Gamma E_S \rho_0)$ 

 $E_{c} = sublimation energy$ 

C = coefficient in Eq. 29, the bulk modulus at low pressures

In the PUFF 66 manual (Ref. 27) a value of 0.25 is suggested for H. The sublimation energy as defined there is the difference between the internal energy of the solid material at ambient conditions and the internal energy of the fully expanded vapor at a temperature of absolute zero.

### (2) Deviatoric Stress

The deviatoric stress equation takes a simpler form in the PUFF formulation than does the pressure equation. The deviatoric stress for a perfectly plastic material is

$$SD = \frac{4}{3} \int_{\rho_0}^{\rho} \mu \frac{d\rho}{\rho} \quad for \quad \left| SD \right| < \frac{2}{3} Y \tag{31}$$

otherwise

$$\left| SD \right| = \frac{2}{3} Y \tag{32}$$

where

# = the shear modulus

p = the density

Y = the yield strength

Stress relaxation and Bauschinger effects modify the deviatoric stress computation. These modifications are dealt with in the following subsections.

# 3. Stress Relaxation Models

### a. Introduction

Materials subjected to shock stresses respond in a manner somewhat different from materials subjected to quasi-static loading. We have attempted to understand and to characterize some aspects of the shock behavior through the use of time-dependent, theoretical models. Attention was focused on materials with no phase change in the stress range being studied. When a shock wave whose amplitude is below a critical overdriving stress traverses such a material, it generally is assumed to proceed initially at the elastic sound speed as a single shock front. As the wave progresses, it breaks into two waves, an elastic precursor and a slower main or plastic wave. The amplitude of the precursor may decrease as the wave proceeds, eventually reaching a steady-state value. The thickness of material through which the shock wave progresses before the precursor reaches its steady-state value varies from material to material. For some materials the thickness is so small (less than 1 mm) that experimental observation of the effect is difficult; for others it is very large.

Three time-dependent phenomena of the shock front were of concern:

- Attenuation of precursor amplitude as a function of distance into a material.
- e Degree of separation of main and precursor wave.
- Attenuation of the stress amplitude of the main wave and modification of other state variables.

It was assumed that these effects are caused by time-dependence of the yield strength or deviatoric stress and that there is no time variation in the relationship between volume and hydrostatic pressure. All of the theoretical models we employed are based on the assumption that the deviatoric stress may exceed two-thirds the static yield strength for brief periods of time. Following the passage of a shock front, the deviatoric stress gradually decays back to its static limit.

The four models are introduced by first considering a model with a sufficiently general form to encompass all of the others. This model is

a specification of the stress relaxing function F in the flow relation presented by Johnson (Ref. 11)

$$\frac{\partial SD}{\partial t} = \frac{4}{3} \frac{\mu}{\rho} \frac{\partial \rho}{\partial t} - \theta F \tag{33}$$

where

SD = the Jeviatoric stress

 $\mu$  = the shear modulus

p = density

 $F = \frac{8}{3} \mu \frac{\partial y}{\partial t}$ , the relaxation function

 $\gamma$  = plastic shear strain

$$\theta = \frac{SD}{|SD|}$$

In this general model F is given by

$$F = A_{1}(\xi) + A_{2}(\xi) (t - t_{a}) + A_{3}(\xi) (t - t_{a})^{2} + \dots$$

$$= 0 \text{ for } t < t_{a}$$
(34)

where

 $\xi = |SD| - \frac{2}{3} Y$ , the excess of deviatoric stress above yielding

Y = the yield strength

 $A_i = functions of \xi$ 

t = time

t = arrival time of the precursor shock front

With the formulation of this model we can now discuss some general results obtainable with all of the models. The devistoric stress may initially exceed 2/3 Y in the shock front and then decay gradually from the high elastic value to the static value. Because of the decreasing deviatoric stress, the amplitude of the elastic precursor also attenuates as the shock wave proceeds into the material. The first term of Eq. 34 provides for attenuation of the precursor. The second and subsequent terms provide for separation of the precursor and main waves, and for details of the wave shapes. The simple ideally plastic model can be represented by the function F with  $A_1 = A_3 = \cdots = 0$  and

$$A_2 = -\frac{4}{3} \frac{\mu}{\rho} \frac{\partial \rho}{\partial t} \frac{1}{t-t}$$
 (5)

Because  $A_1$  is zero there is no precursor decay. With  $A_2$  as in Eq. 35, there is a clear separation of the wave fronts.

Many physical mechanisms have been proposed as a basis for the formulation of the function F. Among these are the standard anelastic solid model (Model 1), several dislocation models, and a varying yield strength. The dislocation models considered here are those of Band (Model 2) and Gilman (Model 3) as modified by Johnson (Ref. 11). The yield strength model (Model 4) was suggested by Anderson, et al (Ref. 3). We will first discuss Model 1, then Model 4, and then turn to Models 2 and 3.

### b. Model 1

The first and simplest "tress relaxation model is that on the standard anelastic solid. It is basically a one-parameter mathematical model in which the deviatoric stress SD is allowed to relax to its static value 2/3 Y exponentially, with an adjustable time constant. Examples of its development are given by Zener (Ref. 2), Kolsky (Ref. 28), Johnson (Ref. 11), and Duvall (Ref. 29). For this model the deviatoric stress satisfied the relations

$$\frac{\partial SD}{\partial t} = \frac{4}{3} \frac{\mu}{\rho} \frac{\partial \rho}{\partial t} - 9 \frac{|SD| - 2/3 \text{ Y}}{T_{\text{plx}}} \text{ for } |SD| \ge 2/3 \text{ Y}$$
 (36)

and

$$\frac{\partial SD}{\partial t} = \frac{4}{3} \frac{\mu}{\rho} \frac{\partial \rho}{\partial t} \qquad \text{for } |SD| \le 2/3 \text{ Y} \tag{37}$$

where  $T_{rlx}$  is the deviatoric stress decay time parameter and the derivatives are taken at Lagrangian coordinates. In Eq. 36 the first term on the right gives the elastic value of the deviatoric stress, and the second term provides for relaxation to the static value 2/3 Y. For this model the yield strength is assumed constant and  $T_{rlx}$  is the only adjustable parameter in the model.

The current value of the deviatoric stress at each coordinate is computed from Eqs. 36 and 37 in the PUFF code. The finite difference approximation to Eq. 36 for the case where  $|SD| \ge 2/3$  Y and dp/dt is constant throughout the time interval is:

$$SD_{f} = \frac{2}{3} \theta Y + (SD_{o} - \frac{1}{3} \theta Y) \exp \left(-\frac{\Delta t}{T_{rlx}}\right) + \frac{4\mu \Delta \rho T_{rlx}}{3 \rho_{avg} \Delta t} \left[1 - \exp\left(-\frac{\Delta T}{T_{rlx}}\right)\right]$$
(38)

where

SD = deviatoric stress at beginning of the time step of duration  $\triangle$  t and density change  $\triangle$  D

SD, = final deviatoric stress

 $p_{\rm BVg}$  = average density during time interval If |SD| becomes less than  $\frac{2}{3}$  Y or changes sign during the time interval, then the SD - t path must be broken into segments. For each segment the correct equation (Eq. 36 or 37) is used.

An alternate physical derivation of the standard anelastic solid can be given from the point of view of dislocation dynamics. For this purpose Eq. 36 is rewritten to emphasize that the third term represents the plastic shear strain rate  $\gamma$ :

$$\frac{\partial SD}{\partial t} = \frac{4}{3} \frac{\mu}{D} \frac{\partial p}{\partial t} - \frac{8}{3} \mu \dot{\gamma} \theta \tag{39}$$

The plastic strain rate is related to the number of mobile dislocations  $N_{m^2}$  the velocity of these dislocations v, and the Burger's vector b by

The model assumes that the dislocation velocity is governed by phonon viscosity, so that

$$v = \frac{b}{\beta} (\tau - \tau_b) \tag{61}$$

where

B = the dislocation damping coefficient

7 = Y/2, the lattice stress that must be exceeded before dislocations can move

T = 3/4 SD | shear stress

By combining Eqs. at through 41 we obtain

$$\frac{\partial SU}{\partial t} = \frac{4}{3} \frac{\mu}{\rho} \frac{\partial \rho}{\partial t} - \frac{2\mu h^2 \chi}{\beta} (SD^2 - \frac{2}{3} \Phi \gamma) \tag{42}$$

Comparison of Eqs. 42 and 36 shows that the two formulations of Model 1 are equivalent if

$$T_{\text{rlx}} = \frac{\beta}{2\mu b^2 N_{\text{m}}} \tag{43}$$

Model 1 is the only one of the four relaxation models for which the precursor amplitude can be obtained analytically in closed form from the various relaxation parameters. From Zener (Ref. 2), the precursor wave amplitude of decays in time according to:

$$\sigma = \sigma_{\text{HEL}} + (\sigma_{\text{o}} - \sigma_{\text{HEL}}) \exp\left(-\frac{At}{T_{\text{rlx}}}\right)$$
 (44)

where

 $\sigma_{\text{HEL}} = \frac{Y}{2} \left( \frac{K}{\mu} + \frac{4}{3} \right), \text{ the Hugoniot elastic limit or final amplitude}$  of the precursor.

$$A = \frac{2\mu/3}{K+4\mu/3}$$

K = bulk modulus

# = shear modulus

Using the relationships

$$\frac{\partial C}{\partial t} = \frac{\partial C}{\partial t} \frac{\partial t}{\partial t}$$
 and  $\frac{\partial C}{\partial t} = \sqrt{\frac{K+4\mu/3}{\rho}} = C_{E1}$ 

which is the elastic shock velocity, the equation for the precursor amplitude decay in distance can be shown to be

$$\sigma = \sigma_{HEL} + (\sigma_o - \sigma_{HEL}) = \exp(-\frac{A^2x}{T_{plx}})$$
 (45)

where

x = distance into the material from the impact interface

$$A' = \frac{2\mu}{3} \sqrt{\frac{(K + 4\mu/3)^3}{(K + 4\mu/3)^3}}$$
 (46)

From either equation, it is evident that decreasing the parameter  $T_{rl}^{-}$  will hasten the decay of the precursor.

### c. Model 4

The next model to be discussed is a two-parameter model similar to that of Anderson et al. (Ref. 13). It is identical to Model 1, except that the yield strength is allowed to vary (the yield strength is constant in Model 1). Under rapid loading, the yield strength reaches a high elastic value and then decays exponentially in time until it reaches its initial value. It is governed by the following equation

$$Y = Y_0 \text{ until } |SD| = \frac{2}{3} Y$$

then

$$\frac{\partial Y}{\partial t} = 2 \frac{\mu}{\rho} \left| \frac{\partial \rho}{\partial t} \right| - \frac{Y - Y_0}{T_y} \quad \text{for } |SD| \ge 2/3 \text{ Y and } SI = N \left( \frac{\partial \rho}{\partial t} \right) = SIGN \text{ (SD)}$$

$$= -\frac{Y - Y_0}{T_y} \quad \text{for } |SD| \ge 2/3 \text{ Y and } SIGN \left( \frac{\partial \rho}{\partial t} \right) = -SIGN \text{ (SD)}$$

$$= 0 \quad \text{for } |SD| \le 2/3 \text{ Y} \quad (47)$$

where

Y = static yield strength

 $T_{v} = yield strength time parameter$ 

Meanwhile, SD obeys Eq. 36 where Y in that equation is now the current value of the yield strength calculated from Eq. 47, instead of the constant value as in Model 1. In Model 4, therefore, there are two adjustable time parameters, one directly governing the deviatents acress decay rate  $(T_{\rm rlx})_{\rm i}$  and another governing the yield strength decay rate  $(T_{\rm rlx})_{\rm i}$ , the latter also indirectly affecting the deviatoric stress decay.

In the computer program, Eqs. 36 and 47 are used to calculate current values of yield strength and deviatoric stress. These equations cannot be solved simultaneously in closed form so we approximate a solution to Eq. 47, for  $|SD| \ge 2/3$  Y, during a time increment  $\Delta$ t

$$Y_{f} = Y_{old} - (Y_{old} - Y_{o}) \frac{t}{T_{y}} + \frac{k \mu \Delta \rho}{\rho} (2 - \frac{\Delta t}{T_{y}})$$
 (48)

where

$$\dot{x}$$
 = a constant =  $\frac{1 \text{ if SIGN}}{0 \text{ if SIGN}} (\Delta p) = \text{SIGN} (SD)$ 

Yold = yield strength value at beginning of time step to

 $Y_f = yield strength value at t = \Delta t + t_0$ 

In this solution we have used the approximation

$$\exp \left(-\frac{\Delta t}{T_y}\right) \cong 1 - \frac{\Delta t}{T_y} \tag{49}$$

which is valid only  $\Delta t \ll T$  (which is the case in all of our PUFF calculations). Inserting Y from Eq. 48 into Eq. 38, we obtain the approximate solution for the deviatoric stress at the end of the time step:

$$SD_{f} = \frac{2}{3} \theta Y_{avg} + (SD_{o} - \frac{2}{3} \theta Y_{avg}) \exp \left(-\frac{\Delta t}{T_{rlx}}\right) + \frac{4\mu \Delta \rho T_{rlx}}{3 \Omega_{avg} \Delta t} \left[1 - \exp\left(-\frac{\Delta t}{T_{rlx}}\right)\right]$$
(50)

where

$$Y_{avg} = \frac{1}{2} (Y_f + Y_{old})$$

Models 1 and 4 are implemented in the subroutine RELAX. The basic scheme of RELAX is as follows: a tentative value for the deviatoric stress is obtained by using the nonrelaxation Eq. 37. This tentative value and the initial value are compared with 2/3 Y to determine if the deviator is outside the elastic zone. If not, the subroutine returns this tentative value as the new deviatoric stress. Otherwise the time during which the deviator remains outside the elastic zone is calculated and this time value is used to relax the yield strength (for Model 4 only) and to calculate a new deviatoric stress using Eq. 38. This new value for the deviator is again compared with 2/3 Y to see if the relaxation has caused the deviatoric stress to regree the elastic zone,

in which case a revised relaxation time is determined and used to recalculate the relaxing quantities. The subroutine then returns with the new value of the deviatoric stress and, for Model 4, the revised value of the yield strength.

The results of a series of representative computations for Model 1 in the SRI PUFF code are shown in Fig. 3. Model 1 yields no distinct separation between the precursor and the main wave until after the precursor has "relaxed" to its Hugoniot elastic limit. As indicated in Fig. 3, there is a gradual rise in stress from the precursor to the main wave peak. The precursor decay curves in Fig. 3 are obtained explicitly using Eq. 41 and they appear to closely match the precursor amplitudes from the PUFF calculations.

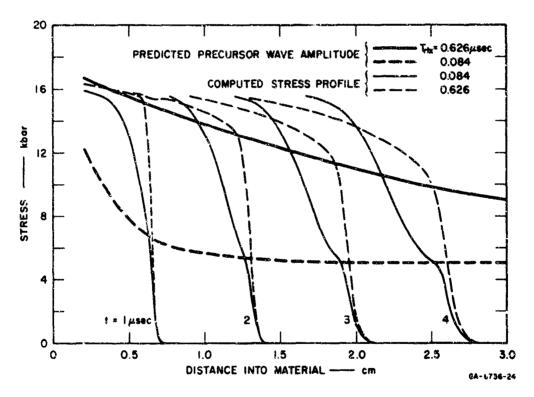


FIGURE 3 STRESS PROFILES FOLLOWING IMPACT IN A MATERIAL REPRESENTED BY MODEL 1 (Simple Anelastic)

The two-parameter variable yield strength model (Model 4) exhibits a distinct two-wave structure, and a noteworthy result of the model is the decrease in stress immediately following the precursor and before the main wave front. Although it has not been possible to derive a closed-form solution for either the precursor amplitude decay or the depth of the trough behind the precursor in terms of the parameters  $T_{rlx}$  and  $T_y$ , the following qualitative information can be obtained from Fig. 4: For a fixed  $T_{rlx}$ , a large  $T_y$  will produce a slowly decaying precursor amplitude and a deep trough behind the precursor. Decreasing  $T_y$  will speed up the precursor decay and decrease the depth of the trough (see curves 2-5 or 6-7 in Fig. 4), until at  $T_y = 0$  (i.e., the yield strength relaxes instantaneously to its static value), the model is identical to Model 1

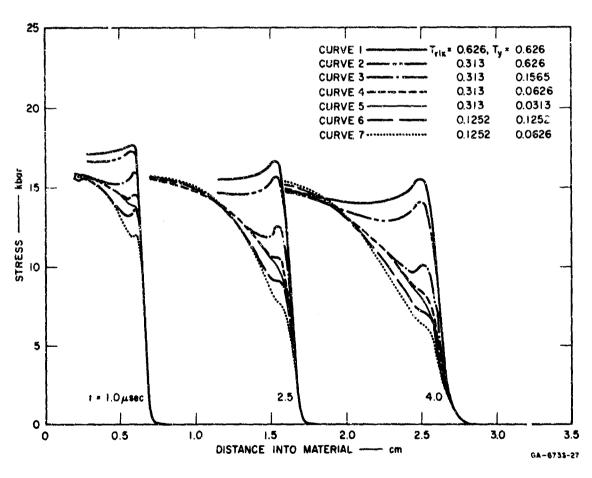


FIGURE 4 STRESS PROFILES FOLLOWING IMPACT IN A MATERIAL REPRESENTED BY MODEL 4 (Two Parameter)

and Eqs. 44 and 45 are valid. If, on the other hand,  $T_y$  is held constant and  $T_{rlx}$  varied (compare curves 1 and 2 or 4 and 7 in Fig. 4), the results show that a decrease in  $T_{rlx}$  will speed up the precursor amplitude decay, but have little or no effect upon the general shape of the shock front. Because no exact equation for precursor attenuation is available, it is not known how much of the precursor decay was caused by the finite difference nature of the calculations.

The two-parameter varying yield model exhibits some of the effects desired of the general two-parameter model described by Eq. 34, but not all of them. In Model 4 the effects of the two parameters are not separable: both affect precursor decay and wave separation. We believe that a more suitable two-parameter model can be generated. The basic purpose of the model is to represent a material with two important stress relaxation mechanisms. In real materials there will generally be a number of mechanisms operating and any two-parameter model will be a simplification of such beh avior, intended to represent only the two most important mechanisms.

#### d. Models 2 and 3

The remaining two models were described by Johnson (Ref. 11) and are based on dislocation dynamics. Model 2 (the modified Band model) and Model 3 (the modified Gilman model) both use Eqs. 39 and 40 of the alternate derivation of Model 1, but then depart from Model 1 when relations governing the dislocation velocity and the number of mobile dislocations are considered.

For dislocation velocity, the following equation applies for both Band and Gilman models:

$$v = v_{m} \exp \left[-B/(\tau - \tau_{b})\right] = v_{m} \exp \left[-B/(\frac{3}{4}|SD| - \frac{1}{2}|Y|)\right] \text{ for } |SD| > 2/3|Y|$$

$$= 0 \text{ for } |SD| \le 2/3|Y|$$
(51)

where

 $v_m = maximum dislocation velocity$ 

B = a constant

The mobile dislocations begin to move when the shear stress exceeds to defined following Eq. 41) and increase in velocity as the shear stress increases. As seen in Eq. 40, the velocity is directly proportional to the relaxation rate of the deviator. From a physical standpoint, the movement of these dislocations relieves the excess shear stress in the material, and the dislocations will continue to move until the shear stress decreases below the lattice strength or until they become pinned.

The differ ace between the Band and Gilman models arises from the manner in which  $N_{\rm m}$ , the number of mobile dislocations per unit volume is calculated (see Johnson (Ref. 11), pp. 27-33, for detailed derivation).

In the Band model, N is obtained from the simultaneous solution of:

$$\frac{dN_{m}}{dt} = g \xi (N_{t} - N_{m}) - (\frac{1}{T} + \frac{1}{T}) N_{m} \frac{v}{v_{m}}$$
 (52.

and

$$\frac{dN_{t}}{dt} = \epsilon_{g} \xi (N_{t} - N_{m}) - \frac{1}{T_{g}} N_{m} \frac{v}{v_{m}}$$
(53)

where

N<sub>t</sub> = total number of dislocations (mobile and pinned) per unit

$$\xi = |SD| - 2/3 Y = \frac{4}{3} (\tau - \tau_b)$$

 $T_1, T_2 = adjustable time constants$ 

€,g = other constants

In Eq. 52 the two terms on the right side represent the rate at which mobile dislocations are created (from pinned dislocations) and destroyed (by being pinned or migrating to a free surface or void), respectively. Similarly, the two terms on the right hand side of Eq. 53 represent the rate at which all dislocations are being created by a Frank-Read multiplication mechanism and destroyed by migration to a free surface or void. To use this model, the initial dislocation densities ( $N_{mo}$  and  $N_{to}$ ) must

be specified, as well as Burger's vector (b), the maximum dislocation velocity  $(v_m)$ , and the adjustable parameters  $(T_1, T_2, \epsilon, g, and B)$ . Then for every cycle of the PUFF - code calculation,  $N_m$  and  $N_t$  are incremented for each cell in the material by:

$$N'_{m} = N_{m} + l_{g} \xi_{avg} (N_{t} - N_{m}) - (\frac{1}{T_{1}} + \frac{1}{T_{2}}) N_{m} \frac{v}{v_{m}} \int \Delta t$$
 (54)

$$N_{t}^{\dagger} = N_{t} + \left[ \epsilon \ g \ \xi_{avg} \left( N_{t} - N_{m} \right) - \frac{1}{T_{p}} N_{m} \frac{v}{v_{m}} \right] \Delta t$$
 (55)

where

 $N_{m}^{t}$  and  $N_{t}^{t}$  = values of  $N_{m}$  and  $N_{t}$  at  $t \approx t_{o} + \Delta t$ 

$$\xi_{\text{avg}} = 1/2 \left| \text{SD'} + \text{SD} \right| - 2/3 \text{ Y} = \left| \text{SD} + \frac{2}{3} \frac{\mu \Delta \rho}{\rho_{\text{avg}}} \right| - 2/3 \text{ Y}$$

$$\approx \xi \left|_{\text{t}} = t_0 + \frac{\Delta t}{2} \right|$$

 $SD^{\dagger} = value of SD at t = t_0 + \Delta t$ 

which is a valid solution to Eqs. 52 and 53 provided that N  $_{m}$  and N  $_{t}$  change by only a small fraction during one time step.

In the Gilman model,  $N_m$  is determined by

$$N_{\rm m} = N_{\rm mo} (1 + C\gamma) \exp (- \emptyset \gamma)$$
 (56)

where C and Ø are constants. The mobile dislocation density is increased in proportion to the total plastic shear strain  $(\gamma)$ , which can be obtained from

$$\gamma = 1/2 \left[ \ln \frac{\rho}{\rho_0} - \frac{3}{4} \frac{(SD - 2/3 Y)}{\mu} \right]$$
 (57)

where  $\rho_{\rm O}$  = initial density (before shock wave). This equation holds only during initial compression. N decreases at a rate proportional to itself, to account for loss by migration to a surface or pinning. To use the Gilman model, one must specify the mobile dislocation density (N<sub>mo</sub>),

Burger's vector (b), the maximum dislocation velocity ( $v_m$ ) and the adjustable constants (B, C, and  $\emptyset$ ). The plastic shear strain is initialized at zero for all cells, and thereafter, whenever  $|SD| \ge 2/3$  Y, the plastic shear strain is increased during a time step by:

$$\Delta \gamma = 1/2 \left| \frac{\Delta \rho}{\rho_{avg}} - \frac{3\Delta SD}{4\mu} \right| = \frac{3}{8\mu} \left| \frac{4}{3} \frac{\mu \Delta \rho}{\rho_{avg}} - \Delta SD \right|$$
 (58)

Note that the term in the parentheses on the far right hand side is just equal to the amount by which the deviator has been relaxed during the time step, since with no relaxation,  $\Delta SD = \frac{4}{3} \frac{\mu \Delta \rho}{\rho}$ , from Eq. 37. Therefore the total plastic shear strain at any time is just equal to  $3/(8\mu)$  times the total stress relaxation in that cell up to that time. This value for  $\gamma$  is used in Eq. 56 to calculate  $N_m$ , which is then used in Eqs. 39 and 40 to obtain the relaxation for the following cycle, which in turn is used to update  $\gamma$ . The accuracy of this iteration is controlled by the shortness of the time step.

Both Models 2 and 3 are implemented in the subroutine BANDRLX, which operates in a manner similar to that of RELAX insofar as the determination of the location of the stress deviator (in elastic or plastic zones) and calculation of the time during which stress relaxation takes place is concerned. Because of the iterative nature of the Band and Gilman schemes, a feature has been instituted in BANDRLX which enables each cycle of the PUFF calculation to be broken into several shorter time steps and a calculation to be made for each of these smaller times, with a resultant increase in accuracy.

The Band and Gilman models are both based on dislocation dynamics, and unfortunately, this is a field in which little is known quantitatively under the loading conditions of interest. There is little or no experimental data for determining the value of the various dislocation parameters (such as  $v_{\rm M}$ ,  $\theta$ , C, g, etc.), some of which like N<sub>to</sub> and N<sub>mo</sub>, undoubtedly depend upon the history of the material specimen. Therefore, there is at this time no sound reason to expect that the Band or Gilman models will describe the shock phenomena more accurately than Model 4, for example, whose parameters are without real physical basis.

The attenuation rate of the precursor can be predicted analytically for these two models, and in fact, the same equation pertains to both.

The equation as derived by Johnson (Ref. 11) as

$$\frac{dS}{dt} = -\frac{4}{3} \mu b v_{m mo}^{N} \exp \left[-B/(\frac{3}{4} SD - 1/2 Y)\right]$$
 (59)

This equation can be integrated numerically to obtain the precursor amplitude. Obviously, only those parameters that enter Eq. 59 can influence the attenuation rate.

Models 2 and 3 exhibit a fairly distinct two-wave shock front, epending to a large extent on the rate of dislocation multiplication in the material. For the case where there is no increase in the number of dislocations (C=0 for the Gilman model, g=0 for the Band model), the shock front is similar to that of Model 1 (see Figs. 5 and 6): a gradual rise in stress from the precursor to the main wave, with no significant change in slope. But when the dislocations are allowed to increase in number (C>0 for the Gilman model, g>0 and  $N_{to}>N_{to}$  for the Band model), the shock front behand the precursor flattens out to some extent before the arrival of the main wave, and thus the two-wave pattern emerges. For the PUFF calculations upon which all of the curves in Figs. 5 and 6 are based, the following values for the various relaxation parameters were used, as suggested by Johnson (Ref. 11). For the Band model:

€ = 0.9

 $T_1 = 1 \mu sec$ 

 $T_{ij} = 4 \mu sec$ 

For both Band and Gilman models:

 $B = 1.98 \times 10^{10} \text{ kbar}$ 

 $v_{\rm m} = 3.22 \times 10^5 \, \rm cm/sec$ 

 $bN_{mo} = 5 \text{ cm}^{-1}$ 

The values for the other parameters are shown in the figures. In addition to the effect of the parameters C and g on the shock front appearance, the graphs show that increasing  $N_{to}$  (in the Band model) will

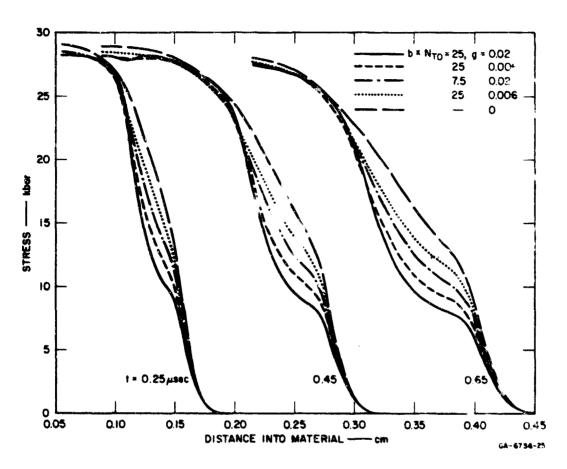


FIGURE 5 STRESS PROPILES FOLLOWING IMPACT IN A MATERIAL REPRESENTED BY MODEL 2 (Band)

increase the separation of the two waves (this is reasonable, since Eqs. 52 and 53 show that the dislocation multiple ation rate is also dependent upon  $N_t = N_m$ ), while an increase in  $\varphi$  (for the Gilman model) will have the opposite effect ( $\varphi$  governs the rate of decrease of the mobile dislocation density). Qualitatively, the Band and Gilman models exhibit a nearly identical shock front appearance.

For both of these models the finite difference calculations have a serious effect on the amplitude of the precursor. According to Johnson's simplified analysis (Ref. 11), C and  $\varphi$  in Gilman's model and N<sub>to</sub>,  $T_1$ ,  $T_2$ ,  $\varepsilon$ , and g in Band's model should have no effect on the precursor decay. In Figs. 5 and 6 it is evident that there is an effect, presumably caused by the artificial viscosity and the finite size of the cells.

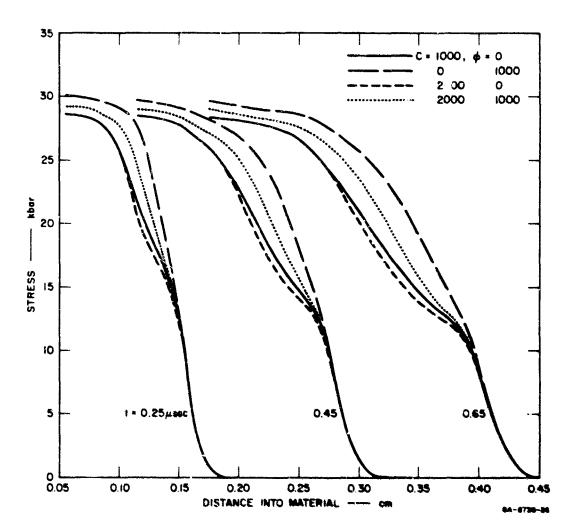


FIGURE 6 STRESS PROFILES FOLLOWING IMPACT IN A MATERIAL REPRESENTED BY MODEL 3 (Gilman)

# e. Implementation in the Code

In the current version of the SRI PUFF code listed in Appendix I, all four stress relaxation models as well as the Bauschinger model (Model 5) are set up to run through the entire cycle of any impact-induced shock wave: compressive loading, unloading, tension, and reloading. The models are also available for use with shock waves caused by radiation deposition, although no verification test run has been made. A condition indicator is assigned to each cell in the PUFF calculation, to keep track of whether that cell is currently outside of the elastic zone (|SD| > 2/3 Y), and hence undergoing relexation, or within

the elastic zone ( $|SD| \le 2/3$  Y), where relaxation is not applicable. The stress relaxation parameters are read in for each material along with all of the other material properties, so that different stress deviator models, including the elastic-plastic model, the Bauschinger model, and all of the relaxation models, can be used in the same calculation for different materials. In addition, two significant parameters for each model are selected and printed out in the periodic time edits and histories, along with the other shock parameters.

The subroutines for calculating the deviatoric stresses for the five models do not contain any COMMON declaration: All required information is brought in as formal parameters in the calling statement. Thus the subroutines are readily transferable to other types of artificial viscosity codes.

For each of the stress relaxation models, a series of test runs has been made with the SRI PUFF code. These runs are identical within each series, except that one or more of the relaxation parameters for that model are varied to determine the effect of the parameter on the calculation.

The results of the test runs show that all four model are similar in that there is an approximately exponential attenuation (in time or distance into the impacted material) of the precursor wave amplitude. But beyond that, there are significant differences among the various models, in the separation between the precursor and the main wave, and generally in the shape of the wave front.

# f. Criteria for Choosing a Stress Relaxation Model

The feature of most prominence in studies of stress relaxation often is precursor attenuation. If this is the feature of main interest the model should be chosen first on the basis of the type of precursor attenuation it provides; that is, first choose the form for  $A_1(\xi)$  in Eq. 34. The form for  $A_1(\xi)$  may be chosen on the basis of measurements of precursor decay or from hypothesized dislocation mechanisms. For the latter criteria, it may be noted that only the variation of dislocation velocity with stress level appears to influence precursor attenuation. Some common forms for  $A_1(\xi)$  are provided in Models 1, 2, and 3.

For Model 1,  $A_1(\xi)$  is proportional to  $\xi = SD - 2/3$  Y. The relaxation time is independent of stress level. In Models 2 and 3,  $A_1(\xi)$  varies as  $e^{-1/\xi}$  so that large stresses decay at a much faster rate than low stresses. Other possible forms for  $A_1(\xi)$  may be found in the literature cited in Section 1.4.

After the form of the precursor attenuation has been selected, then the wave shapes and separation can be considered. They cannot be predicted analytically; therefore, they must be determined by trial computations. We suggest that the first trial should be made for the no-separation case; that is, with  $A_2(\xi) = A_3(\xi) = \dots = 0$  in Eq. 34. This is the automatic form for Model 1. In Band's model this can be accomplished by letting  $\xi = g = N_{to} = 0$  and by making  $T_1$  and  $T_2$  very large. For the same effect in Gilman's model, let  $C = \varphi = 0$ . This first calculation provides a landmark with which to compare the desired results. If these computed results are inadequate, then the arbitrarily set parameters in Band's or Gilman's model can be given more realistic values. Some estimates of the influence of these parameters were presented in the previous discussion of our computed results.

# 4. Bauschinger Model

The Bauschinger effect refers to a change in the stress-strain relations which occurs when the sense of loading is reversed after initial loading. For our purposes the effect causes a continuous variation of yield strength and shear modulus from the point at which loading sense changes to the next change of sense.

# a. Construction of the Model

The recorded stress histories derived from plate impact experiments have usually disagreed with calculations based on the ideally plastic model in three ways: (1) the sharp yield point from the computations is quite gradual in the experimental data, (2) the charp step in the computed unloading of the stress history is almost entirely absent in the experimental records, and (3) the attenuation of peak stress is significantly faster in experimental observation than 1 predicted by computed histories. These discrepancies led us to the formulation of a

Bauschinger model to account for the most significant effects. The model is a strain-rate independent, clastoplastic model with an unloading stress-strain relation that can be very different from the loading relation. Specifically, the model provides for computation of the deviatoric stress (total stress minus pressure) using a shear modulus that varies smoothly from some initial value down to the strain-hardening modulus. The values of initial modulus, strain-hardening modulus and the manner in which the modulus varies between these two are not necessarily the same for loading and unloading.

In constructing the Bauschinger model it was desirable to produce a form that would be general enough to be applicable to many materials and would be readily usable by other investigators. With these needs in mind, the following guidelines were adopted as requirements for development of the model:

- As far as possible, the form should be comparable to those developed by other investigators.
- The form should be extendable to other stress levels.
- The data should be closely represented.
- A physical basis or interpretation should be available for the parameters in the analytical relation.

The data of Buckley and Entwistle (Ref. 19), and the analytical form of Barker (Ref. 23) and of Erkman and Christensen (Ref. 22) were useful in constructing our model. To make the model equations extendable to other stress levels it was necessary to separate the stress-volume relationship into pressure-volume and deviatoric stress-volume equations. Only the deviatoric stress equations were modified by the Bauschinger effect. For selecting the particular equations of unloading stress vs. strain and the loading transition at yielding, we closely followed our data of Section IV for 2024 and 6061 aluminum. The parameters required for the model are mostly moduli and yield strengths and hence are easily interpreted physical quantities.

The pressure volume relationship is given in the usual form for the PUFF code:

$$P = C \mu D \mu^2 + S \mu^3$$
 (60)

where

$$\mu = V_{o}/V - 1$$

The constants C, D, and S are represented by EQSTC, EQSTD, and EQSTS in the code. The stress deviator SD is given by

SD = SD<sub>0</sub> + 
$$\frac{4}{3} \int_{\rho_1}^{\rho_2} M \frac{d\rho}{d\rho}$$
 (61)

where

SD = the value of deviatoric stress at the time corresponding to the lower limit of the integral

M = a shear modulus

If |SD| exceeds 2/3 Y, then |SD| is set to 2/3 Y. The form of the required function of deviatoric stress versus volume is shown in Fig. 7 for 2024-T8 aluminum used on this project. This form was achieved by varying the shear modulus M and yield strength Y according to the following relations:

$$M = M_0 + (M_1 - M_0) \left(\frac{380}{2Y_1}\right)^{N_1}$$
 (loading) (62)

$$Y = Y_1 + 2M_1 \int \frac{d\rho}{\rho}$$
 (loading, after yielding) (63)

$$M = M_2 + (M_3 - M_2) \left(\frac{1}{2} - \frac{3SD}{4Y_2}\right)^2$$
 (unloading) (64)

$$Y = Y_2 + 2M_3 \int \frac{J\rho}{\rho}$$
 (unloading, after yielding) (65)

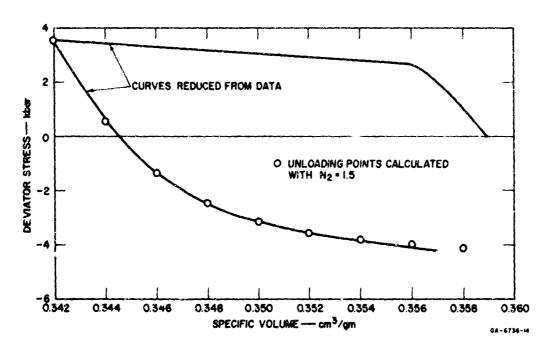


FIGURE 7 RELATIONSHIP BETWEEN DEVIATORIC STRESS AND SPECIFIC VOLUME OBTAINED FROM 50-kbar TESTS IN 2024-T8 ALUMINUM

where

Y<sub>1</sub>, Y<sub>2</sub> = yield strengths at yielding on loading and unloading

Y<sub>2</sub> = the final value of Y from Eq. 63 before unloading begins

M<sub>o</sub>, M<sub>1</sub>, M<sub>2</sub>, M<sub>3</sub> = shear modulus at zero stress on loading, strain-hardening modulus for loading, initial unloading shear modulus, and strain-hardening modulus during unloading

 $N_1$ ,  $N_2$  = constants.

For N =  $\infty$  the relations of ideal plasticity are obtained. Rounded SD-V relations comparable to those obtained from experiments are obtained for N between 1 and 10.

### b. Implementation in the Code

The Bauschinger model is included in the SRI PUFF code as Model 5.

The special parameters and yield parameters used in the model are related to the parameters above as follows:

$$MU = M_{0}$$

$$YADD = 2M_{1}/(\rho_{1} + \rho_{2})$$

$$MUUN = M_{2}$$

$$YADF = M_{3}/M_{1}$$

$$YO = Y_{1}$$

$$XPO = N_{1}$$

$$XP = N_{2}$$

For simplicity in the program and in preparing the data, YADD can be calculated from

$$YADD = \frac{Y_2 - Y_1}{\rho_2 - \rho_1}$$
 (67)

where subscripts 1 and 2 refer to the yield point and beginning of unloading respectively.

The Bauschinger model reflects considerable detailed knowledge about the loading and unloading character of a material. Therefore, it is probably only worthwhile to use the model if the loading equation of state is well known and some information is available about unloading. Then the moduli can be estimated as follows: MU should be known. YADD can be estimated from static data such as Bridgeman's or dynamic results such as Curran's on work hardening. MUUN is the principle parameter determining the rarefaction velocity; hence it should be carefully determined. The rarefaction velocity  $\boldsymbol{c}_{\boldsymbol{R}}$  is

$$c_{R}^{2} = \frac{C + (2D - C\Gamma) \mu_{m} + (3S - \frac{3}{2}D\Gamma) \mu_{m}^{2}}{\rho_{o}} + \frac{\Gamma}{\rho_{m}} (C\mu_{m} + D\mu_{m}^{2} + S\mu_{m}^{3}) + \frac{4}{3} \frac{MUUN}{\rho_{m}}$$
(68)

where

$$\mu_{\rm m} = \frac{\rho_{\rm i}}{\rho_{\rm o}} - 1$$

ρ = density at peak stress
Γ = Grüneisen's ratio (a constant)

and the other parameters are defined following Eq. 60. YAOF may be estimated as 1.0 in the absence of detailed information on the material. The exponent quantities will probably be between 1 and 2 for unloading and about 10 for loading.

Our approach in developing a set of parameters for a material was to calculate as many as possible (i.e., YO, MU, YADD, and MUUN) and then to determine the others by trial. A small computer program was written to compute the stress deviator-volume curves for loading and unloading. The computed values were plotted and compared with the experimental curves. Exponents and YADF parameters were altered until the computed curves were judged satisfactory.

Several computations have been performed with the Bauschinger model to simulate impacts with 2024 and 6061 aluminum. Examples of the output of these calculations are given with the data correlation work of the next section and with the test runs in Appendix I. The results have shown that the model does accurately represent the material behavior. The transition region near the yield point is not required for computations with an artificial viscosity code: the smoothing is caused automatically by the viscosity. Hence XPC should be set  $\geq 100$ . The unloading portion of the calculated stress waves is generally smooth, simulating well the appearance of recorded stress histories.

One defect has appeared in the present model formulation. The fundamental shear modular MU is read in as a constant (which might be obtained from static tensile or compression tests). In reality, MU should vary with density probably somewhat lake the variation in the bulk modulus. In fact, it may be appropriate to write either

$$MU = MU_0 + MU_1 \mu + MU_2 \mu^2$$
 (69)

or

$$MU = (MU/EQSTC) K$$
 (70)

where

K = the effective bulk modulus provided by Eq. 1

MU = the initial shear modulus

 $MU_1$  and  $MU_2$  = constants

Equation 70 reflects the finding of Curran (Ref. 17) that shear modulus appeared to increase in proportion to bulk modulus. Either Eq. 69 or 70 could be used to make a smooth transition from MU to MUUN, the unloading shear modulus. Such a transitic is certainly to be expected on physical grounds in place of the jump in modulus from loading to unloading.

Our work with the Bauschinger model and review of the computed results reveal the great importance of correctly computing the rarefaction velocities. Any attenuation computations made without the Bauschinger model for a material with significant strength may be misleading.

Peak rarefaction velocity, i.e., " ve velocity of the leading edge of the rarefaction fan, as a function of peak stress level can be used to determine the correct form of the MU function. Our review of the rarefaction velocity data presented in Section IV.2.a has shown that MU increases more rapidly with stress than does the bulk modulus. Hence, Eq. 69 may be preferable.

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#### SECTION III

#### EXPERIMENTAL METHODS

# 1. Theory of Data Reduction for Multiple Embedded Gages

An experimental technique has been developed to measure complete loading and unloading paths (in the stress-particle velocity and stress-volume planes) rather than discrete Hugoniot or release points. The shock or compression need not be steady-state and the release may be nonisentropic. The technique, which requires a special experimental configuration and a corresponding data analysis method, is especially suited to examination of yield point phenomena, strain hardening, the Bauschinger effect, and strain-rate effects. The impact experiments employ a target instrumented with a series of stress and/or particle velocity gages embedded at several depths. During the experiment, stress and particle velocity histories are recorded. The data required from the records are the apparent velocities of propagation of stress and particle velocity states from gage to gage. A rigorous derivation of the theoretical basis for the data analysis procedure follows.

In one-dimensional flow the quantities stress, density, and particle velocity are functions of one cartesian coordinate, x, and time, t. The gage analysis is based on apparent wave velocities associated with constant stress or particle velocity states as measured on records from successive gages. We consider two in the analysis: a path of constant stress or particle velocity, and a path with a speed equal to the particle velocity (the path of the gage).

In general, for an observer traveling at velocity  $\xi$  , the total change of  $\sigma$  and u with time will be given by

$$\frac{d\sigma}{dt} = \frac{\partial\sigma}{\partial t} + \xi \frac{\partial\sigma}{\partial x} \tag{71}$$

$$\frac{du}{dt} = \frac{\partial u}{\partial t} + \xi \frac{\partial u}{\partial x} \tag{72}$$

Now consider a path of constant stress in Eq. 71, and a path of constant particle velocity in Eq. 72. In this case  $\xi$  is defined as  $g(\sigma, x, t)$  and W(u, x, t) respectively and the total derivatives in Eqs. 71 and 72 are zero.

$$\frac{d\sigma}{dt} = \frac{\partial\sigma}{\partial t} + U \frac{\partial\sigma}{\partial x} = 0 \tag{73}$$

$$\frac{du}{dt} = \frac{\partial u}{\partial t} + W \frac{\partial u}{\partial x} = 0$$
 (74)

U and W are the velocities of portions of the  $\sigma$  or u pulses with respect to the fixed or Eulerian coordinates. Hence we have a relationship between the partial derivatives with respect to time and space at a point. Now consider the change of stress or particle velocity at the gage, that is, along a particle path. Then  $\xi = u$ , and

$$\frac{d\sigma}{dt} = \frac{\partial\sigma}{\partial t} + u \frac{\partial\sigma}{\partial x} \tag{75}$$

and

$$\frac{du}{dt} = \frac{\partial u}{\partial t} + u \frac{\partial u}{\partial x}$$
 (76)

Substitution of Eq. 73 into 75 and Eq. 74 into 76 yields:

$$\frac{d\sigma}{dt} = (u - U) \frac{\partial\sigma}{\partial x} \tag{77}$$

$$\frac{du}{dt} = (u - W) \frac{\partial u}{\partial x} \tag{78}$$

which apply along a particle path. In order to integrate Eqs. 77 and 78 it is necessary to replace the partial derivatives by total derivatives. This can be done using the equations of motion and continuity.

$$\frac{\partial \sigma}{\partial x} = -\rho \frac{du}{dt} \quad (motion) \tag{79}$$

$$\frac{d\rho}{dt} = -\rho \frac{\partial u}{\partial x} \quad (continuity) \tag{80}$$

where  $\rho$  = density. Substitution of Eqs. 79 and 80 into Eqs. 77 and 78 yields the following set of differential equations:

$$\frac{d\sigma}{dt} = \rho(U - u) \frac{du}{dt}$$
 (81)

and

$$\frac{du}{dt} = \frac{(W - u)}{\rho} = \frac{d\rho}{dt}$$
 (82)

By writing Eqs. 81 and 82 in differential form, using the relation  $V=1/\epsilon$ , and combining the second equation with the first, the following relations between  $\sigma$ , u and V are obtained.

$$d\mathbf{\sigma} = \rho(\mathbf{U} - \mathbf{u}) \ d\mathbf{u} \tag{83}$$

$$du = -\rho(W - u) dV$$
 (84)

$$d\sigma = -\rho^{2}(W - u) (U - u) dV$$
 (65)

The corresponding differential relation for internal energy is derived from the adiabatic relation, dE = - OdV. With the aid of Eq. 85 the relation for energy is

$$dE = \frac{\sigma d\sigma}{\rho^2 (U - u) (W - u)}$$
 (86)

Equations 83 through 85 are the desired  $\sigma-u$ , u-V,  $\sigma-V$ , and  $E-\sigma$  relationships. The next step is to show how the quantities  $\rho(U-u)$  and  $\rho(W-u)$  are derived from gage records. Consider the case of two thin foil stress or particle velocity gages located in the unstressed material at x and x. If the time at which a particular stress level reaches gage i is  $t_i^-(\sigma)$ —then we may define a fictitious (but readily measured) gage velocity  $D_{\sigma}$  as

$$D_{g} = \frac{x_{2} - x_{1}}{t_{2}(\sigma) - t_{1}(\sigma)}$$
 (87)

Similarly, for a particular particle velocity, we define the fictitious gage velocity B.

$$B_{g} * \frac{x - x}{t(u) - t(u)}$$
 (88)

To relate D and B to actual velocities we make use of the conservation of mass between the gages. The mass traversed along any path with slope dx/dt, within the (x-t) plane is given by

$$M = \int_{t}^{t} \rho \left( \frac{dx}{dt} - u \right) dt$$

where dx/dt is a velocity with respect to the coordinates and (dx/dt - u) is velocity relative to the material. In particular, if dx/dt = U, the velocity associated with a stress 0, and t and t are times at which the stress reaches the two gages, then

$$M = \int_{t_{1}}^{t} (\sigma) \rho(U - u) dt$$

Because mass is conserved between the gages,

$$M = \rho_0(x_2 - x_1) = \rho_0\rho_g (t_2 - t_1) = \int_{t_1}^{t_2} \rho(U - u)dt$$
 (89)

where the definition of D , Eq. 87, has also been employed. Evidently  $\rho_{\mbox{o}\mbox{g}}^{\mbox{D}\mbox{g}} \mbox{ is the time average of the quantity } \rho(\mbox{U}-\mbox{u}) \mbox{ between the gages. A similar relationship holds between } \rho_{\mbox{O}\mbox{g}}^{\mbox{B}} \mbox{ and } \rho(\mbox{W}-\mbox{u}).$ 

Because  $\rho_O$  is known and D and B are measurable quantities, we now are able to integrate Eqs. 83 through 86 to obtain the result:

$$u(\sigma, x) = u_{o} + \int_{\sigma_{o}}^{\sigma} \frac{d\sigma'}{\rho_{o} D_{g}(\sigma', x)}$$
(90)

$$V(u,x) = V_0 - \int_0^u \frac{du'}{\rho_0 g(u',x)}$$
 (91)

$$V (\sigma, x) = V_o - \int_{\sigma_o}^{\sigma} \frac{d\sigma'}{\rho_o^2 \frac{D}{g}(\sigma', x) \frac{D}{g}(\sigma', x)}$$
(92)

$$E(\sigma, x) = E_o + \int_{\sigma_o}^{\sigma} \frac{\sigma' d\sigma'}{\rho_o^2 D_g (\sigma', x) B_g (\sigma', x)}$$
(93)

Eqs. 90 through 93 are the relations sought because they provide the  $\sigma - u$ , u - V,  $\sigma - V$ , and  $E - \sigma$  paths during shock and rarefaction regardless of whether the system is steady state or not. As long as suitable gages are available the application of these equations is straightforward, requires no special assumptions, and is capable of determining the entire shock and release paths necessary for computation of propagation and attenuation. In addition the release path measured does not depend on prior knowledge of another material so its accuracy is as good as that of any Hugoniot experiment.

If we consider only simple waves (no wave interactions or rate-dependence) then a line of constant stress is a line of constant particle velocity, and U=W. In this case  $D_{g}=B_{g}$  and the average compression and release paths may be determined with either stress or particle velocity measurements alone. For simple waves the above equations simplify to:

$$u(\sigma) = u_o + \frac{1}{\rho_o} \int_{\sigma_o}^{\sigma} \frac{d\sigma'}{D_g}.$$
 (94)

$$V(u) = V_0 - \frac{1}{\rho_0} \int_{u}^{u} \frac{du'}{B_g}$$
 (95)

$$V(\sigma) = V_{0} - \frac{1}{\rho_{0}^{2}} \int_{0}^{\sigma} \frac{d\sigma'}{p_{g}^{2}}$$
 (96)

$$E(\sigma) = E_o + \frac{1}{\rho_o} \int_{\sigma}^{\sigma} \frac{\sigma' d\sigma'}{\rho_g^2}$$
(97)

The curves defined by Eqs. 90 to 97 do not necessarily lie on the equation of state surface because no steady-state or equilibrium condition is implied by the equations. The computed loading path for a shock follows the Mayleigh line and therefore lies above the equation-of-state surface. If rate-dependence is important, the equations may lead to a series of noncoincident paths (one path for each pair of gages) for the same material.

The foregoing set of equations readily reduce to the more usual forms when the appropriate simplifying assumptions are made. For example in the case of isentropic flow (as in ideal release behavior) the quantity (U = u) in equal to the isentropic sound velocity C. Thus Equation 94 becomes the Riemann integral since in this case  $\rho_0 D \approx \rho (U - u) * \rho C$ .

For steady state shock the wave profile measured by the gage is unchanging in time. Therefore the measured  $\Delta$ t between gage records is constant and hence D = B = constant. Since in general D =  $\frac{\rho}{\rho_0}$  (U - u),

we can evaluate D at zero stress and D is therefore equal to  $(U-u_0)$ . By making this substitution Eqs. 94 and 94 implify to the equations for the Rayleigh line in the U-u and U-V planes respectively. If in this simple case Eqs. 94, 95 and 96 are integrated across the shock front, we obtain:

$$\sigma - \sigma_{c} = \rho_{o}(U - u) (u - u_{o})$$
 (98)

$$1 - \rho_0/\rho = \frac{u - u_0}{U - u_0} \tag{99}$$

$$\sigma - \sigma_{o} = -\rho_{o}^{2}(v - v_{o})^{2}(v - v_{o})$$
 (100)

Integration of Eq. 97 and simplification with the aid of Eq. 100 leads to

$$E - E_0 = -\frac{1}{2} (\sigma + C_0) (V - V_0)$$
 (101)

Equations 98, 99, and 101 are the usual Hugoniot relations.\*

A sample application of the multiple-gage analysis procedure is presented in Section IV for aluminum.

<sup>\*</sup>Peterson et al., to be published in Geophysical Research.

### 2. Experimental Techniques

### a. Gas Gun Experimental Techniques

The SRI light gas gun described by Linde and Schmidt (Ref. 30) is used for impacting samples with flying plates (thin-headed projectiles) to produce plane stress shock waves below 100 kbar traveling into both the target and flyer plate. The projectile material and velocity may be selected for each experiment so that, for example, the projectile head may be of the same material as the target. This "symmetric impact" configuration is often used because it offers a guarantee that the particle velocity in the target at the impact interface will be exactly one half the projectile velocity. If the target exhibits steady state flow, its Hugoniot state is usually determined by measuring the projectile velocity (and thus particle velocity) and the shock velocity in the target and by applying the Rankine-Hugoniot conservation equations (jump conditions) to calculate the other variables.

Figure 8 illustrates the experimental arrangement used in the present study to investigate loading and attenuation in aluminum and titanium alloy at stresses below 50 kbar. The flyer plate may be bonded directly to the projectile or may be supported by a foam material in front of the projectile as in Fig. 8. With the foam backing, the plate behaves approximately as a free flyer and therefore can be used in attenuation studies. The bonded plate configurations are used for shots where only loading is of interest.

When very thin flyer plates were required, a foam backing surrounded by a tubular aluminum extension was added to the projectile to support an aluminum flyer plate (disk) at its circumference. The cavity behind the flyer was filled with a rigid, low-density, high-strength polyurothane foam to ensure that the flyer remained flat while the projectile was accelerated. Tests using an optical flat and a monochromatic light source showed that the foam did not expand when subjected to typical barrel vacuums and the flyers remained flat to better than three microns. The shock impedance of this foam could be neglected compared to that of aluminum although this assumption was not required in the data analysis.

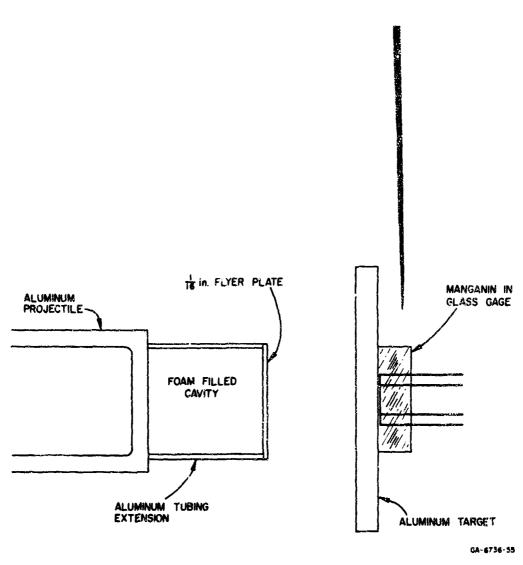


FIGURE 8 EXPERIMENTAL ARRANGEMENT FOR GAS GUN ATTENUATION SHOTS

The stress history at the back face of the metallic specimens was measured by means of manganin transducers embedded in glass. For the shots with aluminum targets, a soda-lime glass was used with the transducers because this glass has a shock impedance only slightly below that of the aluminum alloys investigated. Therefore, to s good approximation, wave interactions caused by reflections at the aluminum-glass interface could be ignored. To minimize target-gage wave interactions in the tests with titanium, a lead oxide glass was incorporated in the transducers in place of the soda-lime glass. This glass has an impedance only slightly lower than that of titanium. The manganin transducers are discussed further in Section III.3.

# b. High-Explosive Experimental Techniques

In the present study, specimen stresses between approximately 100 and 700 kbar were generated with high-explosive (HE) driver systems. The three driver systems employed (Table 1) are of one or the other of the two types shown in Fig. 9. The first system propels a thin, high-velocity flyer for attenuation studies. The second, and simpler, type of HE system is used in experiments which require measurement of the loading state but not of attenuation or other unloading properties.

Both types of system employ an 8-inch-diameter plane wave generator (P-80) to initiate a plane detonation wave in a cylindrical pad of HE, which either directly or indirectly (see Fig. ?) accelerates the flyer across the flight path to the target. Specimen shock state measurements are made electronically with manganin gages or optically with a smear camera or with a combination of these two techniques.

The first HE system (Fig. 9, System E) was developed by Erkman and Christensen (Ref. 22) to accelerate thin, high-velocity, relatively stress-free, unspalled flyers. Such flyers are required to study shock attenuation and stress release in shocked material resulting from the trailing rarefaction fan originating at the back surface of the flyer (see Fig. 10) Christensen found that when the plane wave generator was in direct contact with an 1/8 in. thick aluminum flyer, the flyer would spall during the experiment. To avoid flyer spalling, the air gap, buffer plate,

Table 1

PARAMETERS OF EXPLOSIV'LY LAUNCHED FLYER PLATE SYSTEMS

* Explosive Flyer System	Driver Free-Surface Velocity Velocity V	Driver Shock Stress <sup>†</sup> O (kbar)	Particle Velocity <sup>†</sup> u (mm/µsec)
System B	4.8 ± 0.1	550±20	2.3±0.1
System D  p-80 + 2 in. Baratol + 1/16  in. air gap + 1/2 in. 2024 Al  flyer + 1 in. flight + 3/16 in.  2024 Al driver§	2.2 ± 0.1	200±10	1.1±0.1
<pre>System E P-80* + 2 in. Comp. B + 1/2 in. air gap + 3/4 in. Al buffer** + 1/8 in. sheet explosive + 1/8 in. 2924 Al flyer + 1 in. flight + 3/16 in. 2024 Al driver§</pre>	3.1 ± 0.1	310±15	1.5±0.1

All dimensions are uniform within  $\pm$  0.001 inch and so (aces of all components are flat and parallel within  $\pm$  0.0005 inch.

The tolerances denote the range within which approximately 80 percent of the measurements fall.

Plane wave generator 8 inches in diameter.

The driver thickness and material were varied in some of the experiments.

In one experiment the buffer thickness was 1/2 inch.

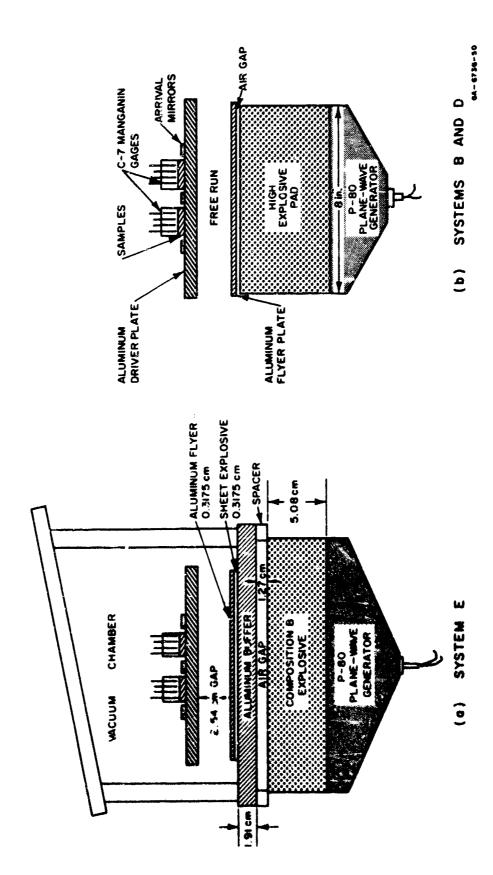


FIGURE 9 SIDE VIEW OF HE FLYER SYSTEMS B, D, AND E FOR GENERATING UNIAXIAL STRAIN LOADING AND UNLOADING STATES

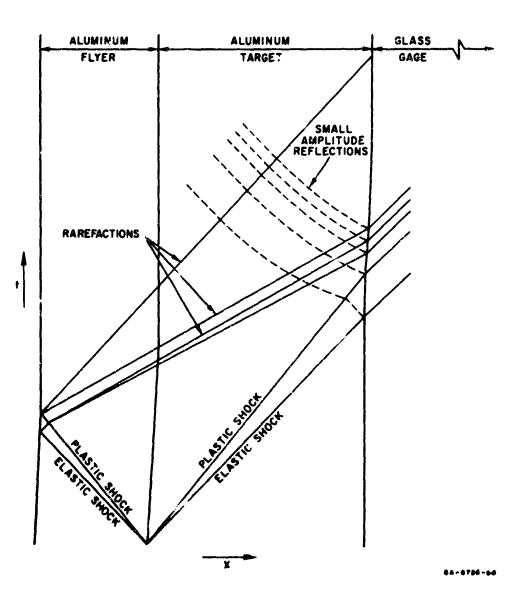


FIGURE 10 A DISTANCE (x)-TIME (t) DIAGRAM SHOWING SHOCK AND RAREFACTION WAVES CAUSED BY IMPACT

and sheet explosive (providing supporting stress at the back of the flyer) were added by Christensen and Erkman (Ref. 22). The resulting flyer velocity is approximately 3 mm/µsec, with planarity (simultaneity of arrival) of 0.04 to 0.06 µsec across a 4-inch-diameter impact surface. The vacuum system shown in Fig. 9 is employed to eliminate the air shock which would otherwise precede the impact. A typical experimental configuration using System E is shown in Fig. 11.

The second type of HE system, represented here by Systems B and D (Fig. 9), is characterized by the fact that the high-pressure detonation products act directly on the flyer (usually much thicker than flyers of System E) to accelerate it across the free run gap into the target. Systems B and D were demonstrated not to spall by letting the flying plates impact manganin stress transducers and examining the resulting stress histories. The planarity of the shock front in the target obtained using these systems is of the same order of magnitude as that of System E.

The objectives of the HE experiments in this study were to measure Hugoniot states achieved in the Specimens, and in the case of the thin Elyer attenuation shots, to record the stress history during loading and unloading. In all the HE experiments Hugoniot states were determined using optical techniques (Ref. 31,32), and stress histories were obtained electronically using manganin C-7 stress transducers (Ref. 33).

Hugoniot state determinations in the HE experiments were simplified by the observation that the study materials (with the exception of the three-dimensional weave material) supported a single-front, steady-state shock. Therefore, the Eunkine-Hugoniot conservation equations apply and the impedance match technique (Ref. 34) is valid. Under these conditions measurement of two shock variables allows the determination of the specimen Hugoniot state. In the present experiments the two variables measured were the driver free-surface velocity (or equivalently the flyer velocity) and the specimen shock velocity. Since these velocities are constant during the experiment (steady-state flow), they were determined from time and distance measurements. Back surface mirrors were positioned

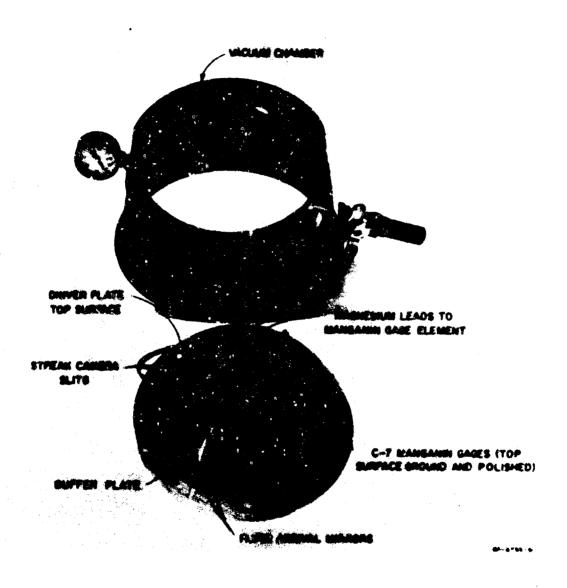


FIGURE 11 SYMMETRICAL IMPACT EXPERIMENT (HE System E) TO DETERMINE HUGORIOT DATA AND SHOCK WAVE PROFILES

at the planes of interest (e.g., the front and back surface of the specimen) and illuminated by a relatively long-duration, shocked argon light bomb. These mirrors were monitored by a time-resolving high-speed streak camera which recorded the relative times of the events of interest (e.g., the shock front arrival at the front and back surfaces of the specimen) by the very rapid (occurring in < 10<sup>-8</sup> sec) changes in mirror reflectivities that resulted. With such optical methods velocities can be determined to within 1 percent in a typical experiment.

To determine stress histories in HE experiments, manganin/C-7 gages (Ref.33) were mounted on the specimen. These gages record the entire stress history in C-7 resulting from the shock transmitted across the specimen C-7 interface. Ideally these experiments would employ a transducer medium which closely matched the shock impedance of the specimen, to minimize the waves reflected back into the specimen from the specimen-transducer interface (see Fig. 10). However, manganin/C-7 gages were used in the HE experiments (in spite of being of lower shock impedance than the study materials) because the manganin/C-7 gage is the only one which has been calibrated at these high stresses. It is therefore fortunate that in all HE experiments the study materials support only a single front shock (no elastic precursor or phase transition wave), so interpretations of the gage experiments are not complicated by interactions in the specimen between the rarefaction reflected from the interface and a complex plastic shock front. However, in the region where the rarefacted waves reflected from the gage-target interface intersect the rarefacted waves from the free surface of the flyer (Fig. 10), the siress will become tensile and the specimen may spall. The stress release seen in C-7 is therefore characteristic of the interaction of the two rarefaction fans.

In all RE experiments the transducers were standard 4-terminal 1-ohm gages (Ref.33) with the manganin element (0.002-inch-diameter wire flattened to less than 0.001 inch to minimize response time) set 0.010 inch into the C-7 from the specimen/C-7 interface. It was noted that when this particular manganin/C-7 gage configuration was shocked to ~745 kbar peak pressure in C-7, a 60 to 70 mw signal, commonly lermed a polarization signal, occurred.

The signal appeared 30 to 40 nsec before the main pulse, corresponding to the shock transit time through 0.008 to 0.012 inch of C-7, dropped to 10 to 20 mg as the shock entered the manganin element, and decayed to zero approximately 1/2 μ/cc later. The signal is believed to be generated by the leakage of induced charge in the C-7 through the active manganin element to ground. In the data analysis this signal was subtracted from the transducer record to obtain the true stress history. To compensate the gage hysteresis during stress release, the observed relative resistance change of the mang lin wire is assumed to depend linearly on axial stress (as during loading) but with the proportionality constant decreased during unloading by 12-1/2 percent for peak stresses between 130 and 160 kbar (Ref. 33). Thus when ΔR/R, the relative resistance change, has decreased to 12-1/2 percent of its peak value, it is assumed that the axial stress has been completely released.

The shock front arrival and planarity at the transducer-specimen interface was monitored optically by locking through the ground and polished top surface of the C-7 transducer at the polished specimen below using the streak camera. For the study of nonmetallic materials, the gage front surface was aluminized. This combined electronic-optical instrumentation proved highly successful and is recommended when detailed shock planarity information is needed in conjunction with stress transducer information.

### c. Manganin-in-glass Gages

The manganin-glass gage consisted of four terminal  $\eta$  network manganin elements (manganin ribbon 0.0005-inch thick and 0.025-inch wide was used) with mercury-silver dental amalgam leads made by back filling 1/16-inch noies in 1/2-inch-thick glass gage blocks. Aluminum foil disks (0.0004 inch thick by 3/16 inch diameter) were pressed onto the dental amalgam-manganin ribbon juncture and a 0.003-inch-thick cover glass was bonded over the face with C-7 epoxy. The aluminum foil disks extend

<sup>\*</sup> C-7 resin with 6 percent activator A by weight, from Armstrong Products, Inc., Warsaw, Indiana.

the recording time (typicarly 2 usec) of these transducers by maintaining good electrical contact at the ribbon-load joint during snock loading.

Kanganin transducers in soda-line glass were calibrated for perk stresses up to 43 kbar by impacting test glass gages incorporating manganin ribbon elements at two different depths, with either Impactor or aluminum flyers. Measurement of shock transit time between the gager yields the shock velocity in the glass. Since the Hugoriot of the projectile material was known, the paak stress could be determined by the impedance match method. The piezoresistive response measured in these tests was linear and equal to 0.9029 N/N/kbar, identical to that previously obtained in Ce7 epexy.

The Hugoniot and unloading characteristics of the soda-lize glass were also determined from these experiments. The resistance-time records generated by place impacts revealed that the stress wave in glass was steady state for both the compression and release portions and that the shock velocity was constant up to 43 kbar. The conclusion was that the Hugoniot and the unloading isentrope are a single straight line in the stress-particle velocity lane. The glass shock impedance measured in these experiments was  $\rho_{\lambda}U = 14.9 \pm 0.1$  g/cm<sup>3</sup> mm/ $\mu$ sec. It is interesting to compare this result with that of Fuller and Price (Ref. 35) who investigated a soda-lime glass with a density of 2.46 gm/cm3. The shock impedance of their glass was some 10 percent lower than that of the glass used in this work and their glass displayed strong dispersion of the compressive wave. This indicates that density is not an adequate indicator of dynamic properties in different sods-lime glass. Indeed, discussions with glass suppliers revealed that the composition of soda-lime glass can vary considerably from batch to batch, even from the same supplier. Consequently, all glass used in this work was taken from a single sheet, with the exception of the 0.003-inch-thick cover glass.

Soda-lime glass ( $\frac{1}{2}$ -inch-thick plate window glass),  $\rho_0 = 2.52$  g/cc.

\*\* A polycrystalline alumina (Al $_2$ O $_3$ ) produced by General Electric Co., Lamp Glass Department, Los Angeles, California.

Gage hysteresis was measured by impacting test gages with Lucalox flyers. The impedance mismatch between glass and Lucalox causes a series of reverberations in the Lucalox flyer. If all stresses are below the elastic limit of Lucalox, the reverberating waves remain nondispersive and the glass release states are well-defined, thus providing a release calibration of the manganin in glass transducer. The results of release calibrations for peak stress up to 43 kbar are summarized in Fig. 12 which shows that the percentage drop in gage resistance lags the percentage drop in stress during release; i.e., the gage exhibits considerable hysteresis. The recorded stress histories were corrected for release hysteresis according to this curve.

For tests with titanium a lead oxide glass called Hi-D was used. The shock impedance of Hi-D is slightly lower than that of titanium. Hugoniot measurements (Ref.36) on Hi-D glass have been performed in this laboratory, and the glass used in these gages came from the same source as that used for the Hugoniot measurements. The Hi-D glass Hugoniot is a straight line up to 150 kbar with  $\rho$  U = 20.7 g/cm<sup>3</sup> mm/ $\mu$ sec. The previously measured piezoresistive coefficient of the manganin in Hi-D glass transducer, is 0.0024  $\Omega/\Omega$  kbar (Ref.37).

<sup>&</sup>lt;sup>\*</sup> 89 percent PbO glass,  $\rho_0$  = 6.2 supplied by Penberthy Instrument Company, Seattle, Washington.

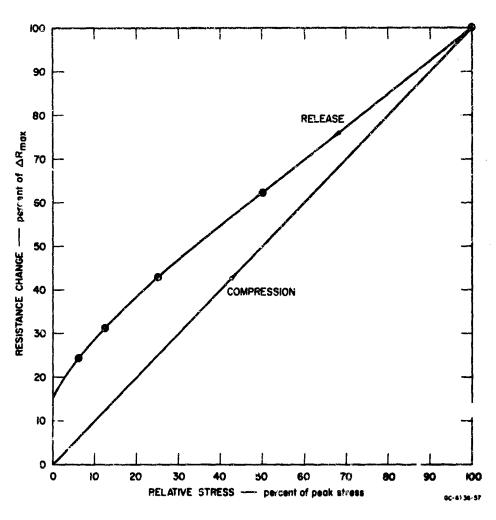


FIGURE 12 MEASURED UNLOADING CALIBRATION OF MANGANIN ELEMENT IN SODA-LIME CLASS

AFWL-TR-69-96

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### SECTION IV

### STUDY OF ALUMINUM

### 1. Introduction

For the study of aluminum the objectives were to investigate stress relaxation, unloading effects, and stress attenuation. The design of the experiments was based on the data analysis technique outlined in Section III, a technique which was still being developed. Aluminum was a particularly good choice for the experimental program because many of its properties are well known, and even its unloading characteristics have been studied (Erkman, Christensen, and Fowles (Ref.22) and Barker (Ref.23)). Stress relaxation effects have been observed in some aluminum alloys. Based on results of experiments at low strain rates, it was expected that 2024-T8 would be strain-rate-dependent while 6061-T6 would be relatively rate-independent. Another reason for choosing aluminum was its high experimental reproducibility, a property of special importance in any basic investigation.

The two aluminum alloys selected for study, 6061-T6 and 2024-T8, were supplied by Kaiser Aluminum Company, Department of Metals Research, Spokane, Washington. All sheets of each alloy (from 1/16 inch to 1-1/4 inches thickness) were supplied from a single billet to guarantee uniformity. The heat treatments, ageing, and rolling work hardening were those specified by the standards for T6 and T8 (Ref. 38, 39). Table 2 presents a summery of the alloy properties obtained from quasi-static and acoustic measurements. A description of these tests and detailed results are given in Appendix II.

The dynamic study of the two aluminum alloys consisted of:

- e Fifteen to 20 experiments on each alloy at four or five nominal pressure level e between 15 and 500 kbar, including two series of impact experiments at atress levels of 50 and 300 kbar with specimens of various thicknesses.
- Reduction of the experimental data to obtain a complete stress-volume relation for loading and unloading.
- Wave propagation calculations to correlate with the experimental stress records.

Table 2

MEASURED STATIC PROPERTIES OF ALUMINUM ALLOYS

Alloy	Density O (g/cm <sup>3</sup> )	Young's Modulus, E (kbar)	Vield Strength (uniaxial tension) (kbar)	Ultimate Strength (uniaxial tension) (kbar)	Poisson's Ratio, V
2024-T8	2.785	745	4.4	4.9	0.33-0.34
6061-Т6	2.709	710	2.8	3.1	0.32-0.34

Notes: Densities are from a series of measurements on each alloy.

Moduli and the two strengths are from quasi-static measurements reported in Appendix II. Poisson's ratio is from both acoustic and quasi-static measurements of Appendix II.

### 2. Dynamic Experiments

Both attenuation and Hugoniot-type impact tests were conducted on the two aluminum alloys, 2024-T8 and 6061-T6. The gas gun experiments, which provided considerable detailed information about precursor and rarefaction waves, are described first; then the HE (high explosive) experiments are described. The experimental configurations and the Rugoniot data obtained are summarized in Tables 3 and 4 and Figs. 13 and 14.

# a. Gas Gun Tests

Seventeen gas gun experiments were performed on 2024-T8 and 6061-T6 aluminum using 1/16-inch-thick flyers to examine attenuation in these materials. Four gas gun experiments were performed on 6061-T6 with 1/8-inch-thick flyers to examine shock propagation without attenuation in thick targets. Targets for both sets of experiments were nominally 1/8, 1/4, 3/8, or 1/2 inch thick. The experimental configuration is shown in Fig. 8. A typical set of manganin stress transducer resistance-time records for four different target thicknesses is shown in Fig. 15. (The ratio of target to flyer thickness is given as R in the figure.) The blanking marks on these traces are used for time correlation of stress

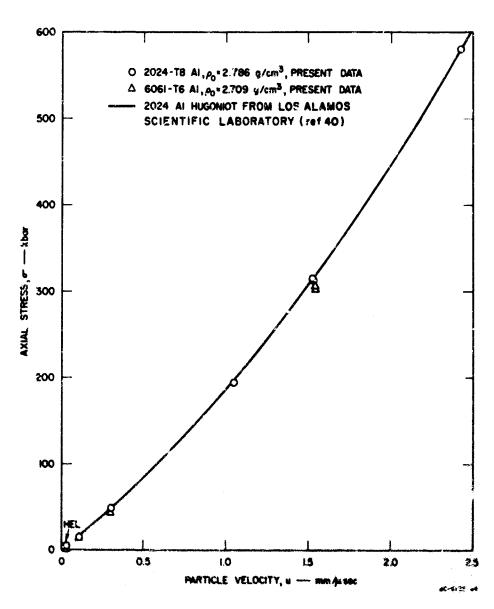


FIGURE 13 - STRESS-PARTICLE VELOCITY HUGONIOT DATA FOR 2024-TB AND 8081-TB ALUMINUM

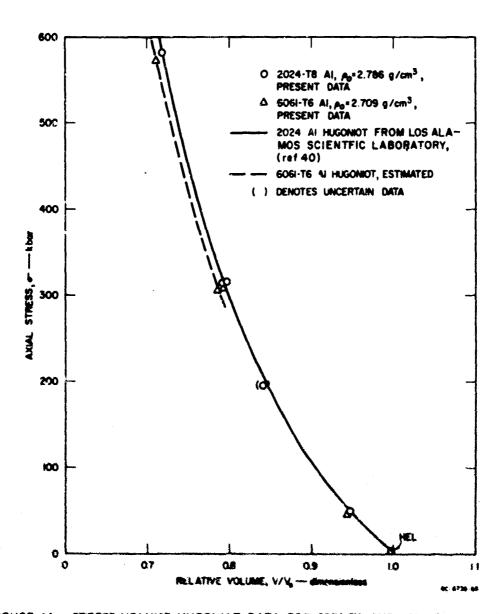


FIGURE 14. STRESS-VOLUME HUGONIGT DATA FOR 2024-T8 AND 8081-T6 ALUMINUM

Table 3

The state of the state of

Compression of the control of the co

NUMBER OF THE STATE ALBEITHM.\*

	_	Driver Syst	100		Bug	Magoniot Elestic Limit	tic Limi	£‡	712	Final (Magoniot) State	lot) Sta	•
i de la companya de l	Byeto	Flyor* Flyor (am/hose) (am/hose)		Specimen Thickness (see)	Specimes Shock Particle Axial Specific Thickness Velocity, Velocity, Stress, Volume (see) (see) (wm/Lsec) (kber) (cd²/g)	Shock Particle Axial Specific Shock Particle Axial Velocity, Velocity, Velocity, Velocity, Velocity, Velocity, Velocity, Velocity, Velocity, Stress, U U U U U U U U U U U U U U U U U U	Axial Stress, g (kber)	I	Shock Velocity, U (am/usec)	Shock Particle Axial Specific Velocity, Velocity, Stress, Volume, U U G G (am/usec) (kbar) (cm²/g)	Axial Stress, G (kbar)	Specific Volume, V (cm <sup>2</sup> /g)
2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2		0.214 0.218 0.214 0.217	****	3,36 8,76 12,56	.2	6.025°	* *	0.3874*	w/y*	0,108*	17.0*	<b>‡</b>
		\$ 1 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	1211177	3 2 3 3 3 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4		0.026	*5.	0.3674	ø	0.303*	*0.	o. 34.80
13,530	( H	30.5	1		1	1	1	;	7.26	1.55	30,	0.280
13,560	<u></u>	3,03	!	6.21	;	1	ı	i	7.38	2.2	308	0.292
13, 527 B(ME)	<b>(11)</b>	4.85	***	4.8	•	•	•		9.36	2.46	572	0,263

Average initial density = 2.700 g/cs with specimen-to-specimen variation of ±0.005 g/cm2. Asterisks (\*) indicate averaged values. S and U, u, and V were obtained from Figs. 19, 14, 30, and 21, respectively.

Magnatot elastic limit (MEL) is dofined in the text. No MEL was observed in the ME experiments.

Projectile velocity for gam system; flyer velocity or driver free-surface velocity for ME system.

Dispersive plastic eave; so unique shock velocity.

Attendating wave; so Magnalot state,

Table 4

# MINDOMICT DATA FOR 2024-TS ALUMINUM

		Driver Syste			<b>1</b>	Magnator Clastic Limit	tic Limi	£	Pte	Final (Magoniot) State	ot) Sta	•
ij		Priver Priver Printmess (mm/kinec) (mm)		Specials Shock Particle Axial Specific Shock Particle Axial Specific Thickness Velocity, Velocity, Stress, Volume, a U U U U U U U U U U U U U U U U U U	Shock Velocity, U (ma/keec)	Shock Particle Axial Specific Shock Particle Axial Yelocity, Velocity, Stress, Volume, Velocity, Velocity, Stress, U U U U U U U U U U U U U U U U U U	Azial Stress. G	Specific Volume, V ( on 7/8)	Shock Velocity, U (mm/wsec)	Particle Axial Specific Velocity, Stress, Volume, U O V V V V V V V V V V V V V V V V V V	Axial Stress, o (kbar)	Specific Volume, V (cm <sup>2</sup> /g)
8 8	33	0.236	32.	3.18 }	•	0.033	0.0 0.0	0.3570	\$	0.10	17.5	<b>‡</b>
9 9 9 9 9 11 to 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5		0.410		20.00	*s.	0.031	*0.	0.3873		0.30	***	0.340
13, 526	( <b>EE</b> )	1.00		3.	1	}	1	1	6.674	1.05**	195**	0,302*
13, 550 <sup>64</sup> E(HF	E ( 85	3.03	!	• • •	ı	1	1	ł	7.36	1.53	314	0.284
13, 532	) (M (M (M (M (M (M (M (M (M (M (M (M (M (	3.03 80.2	1 1	**	! ! !	11	1 1	1 1	7.44	1.52	316	0.286
13, 527	B(.XXX.)	4.25	-	4.72	1	:	••	•	8.65	2.43	583	0.258

Average initial density \* 2,786 g/cm with apecimen-to-apecimen variation of ±0,005 g/cm. Asteriaks (\*) indicate averaged values. O and 1, u, and Y were obtained from Figs. 13, 14, 20, and 21, respectively.

Magosiot slastic lisit (MNL) is defined in the text. No MRL was observed in the ME experiments.

Projectile velocity for gan system; thyer velocity or driver free-surface velocity for HE systems.

Dispersive pissiic wave; to unique shock velocity.

Attenuating wave; no Magoniot state,

Value not calculated.

Mirror cutout time was not sharply defined on smear camera record; hence, precision was low.

Specimen mounted on 4.86-mm-thick driver plate of 6061-76.

### ALUMINUM (6061-T6)

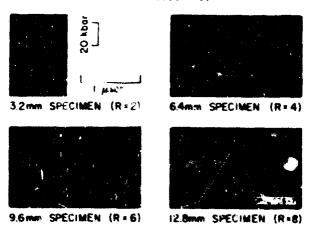


FIGURE 15 TYPICAL MANGANIN TRANSDUCER
RECORDS OBTAINED FROM:
IMPACTS IN 6081-T8 ALUMINUM

gage data with impact time data. A more precise time base is provided by a separate 10 MHz time mark trace (not shown). Aluminum-glass interface stress-time profiles obtained from transducer records by correcting for gage hysteresis are shown in figs. 16 and 17. Because of the good impedance match between aluminum and the glass transducers, the effects of reflections from the aluminum-glass interface on later portions of the incoming wave three neglected for the multiple-gage analysis used to compute the continuous stress-volume relations. (This assumption was not made in calculating Hugeaiot points nor in the wave propagation computation of the SRI-PUFF code.) The adequacy of the impedance match assumption was checked experimentally with an "in-material" gage in one experiment. In that test a manganin foil gage (0.0002-inch-thick) was assistiched between two layers of aluminum, using 0.001-inch-thick mica as hasulation on each side of the foil (thus producing an "in-material" gage). It is expected that this sandwiched gage disturbs the flow field very slightly, but it is not known how well the gage response follows interface stresses during unloading. The record from this "in-material" gage is in general agreement with the glass transducer record as shown in Fig. 18, although there are differences, particularly in the unloading portion of the wave. These differences show that the reflections from the aluminus-glass interface may not be entirely negligible.

The error bounds of measured impact time were excessive in some experiments due either to tilt or distortion of the flyer. Improvement in the time correlation of the stress-time profiles of Figs. 16 and 17

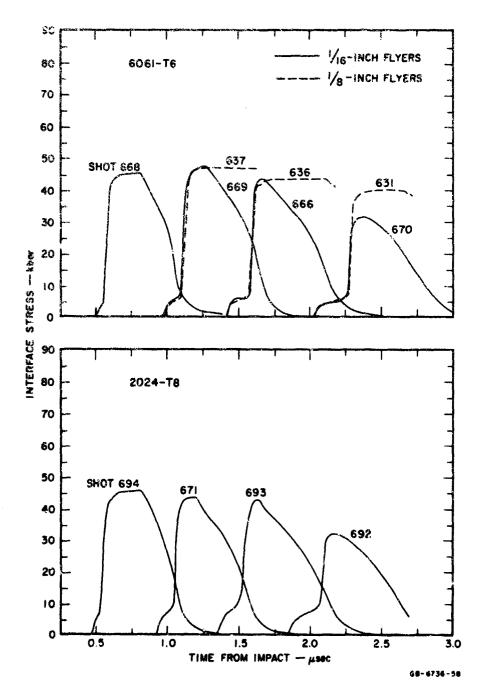


FIGURE 1" INTERFACE STRESS HISOTRIES RECORDED BY MANGAIN-GLASS BEHIND ALUMINUM TARGETS (50 kbar Impacts)

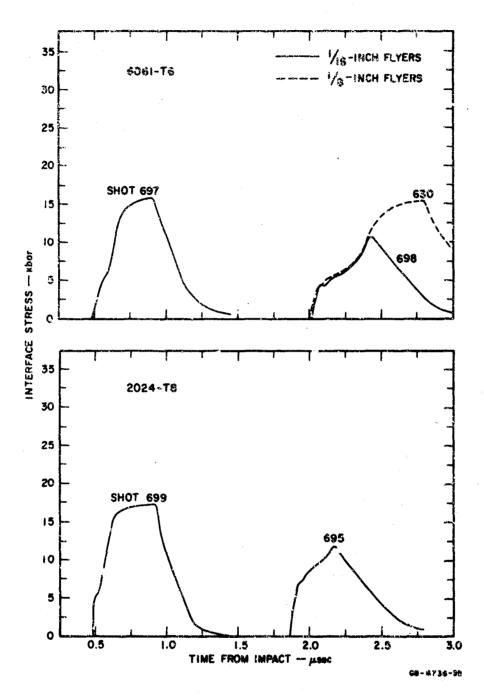


FIGURE 17 INTERFACE STRESS HISTORIES RECORDED BY MANGANIN-GLASS GAGES BEHIND ALUMINUM TARGETS (20 kbar impacts)

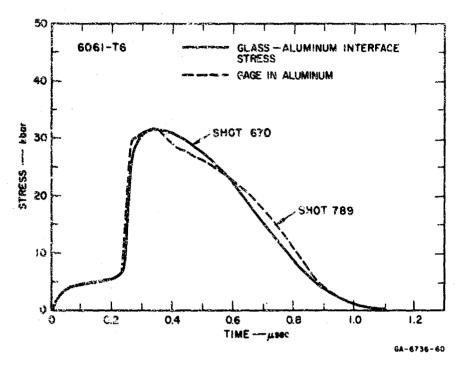


FIGURE 18 COMPARISON OF STRESS HISTORIES RECORDED
AT ALUMINUM-GLASS AND ALUMINUM-ALUMINUM
INTERFACES

was achieved by assuming that the steady-state portions of the wave, both elastic and plastic, were propagating at constant (but, of course different) velocities. The positioning of the profiles in time by this method gave elastic wave velocities that were in good agreement with the longitudinal sound velocity measurements.

Average values of the product  $\rho$  D, (which we shall call instantaneous impedance) as a function of stress were obtained from these profiles (as described in Section III.1) for both compressive and release waves and are plotted in Fig. 19. Averaging was accomplished by measuring  $\rho$  D as a function of stress for every combination of profiles in Figs. 16 and 17 and then weighting each  $\rho$  D value according to the initial distance between gages. This weighting procedure was used since the measurement error was a fixed time increment, and therefore the percentage error should be smaller for the larger separations. These ata then provide the required information to integrate the momentum relations

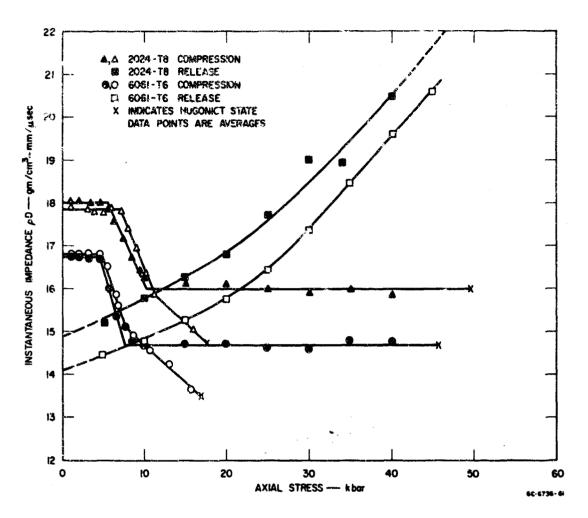


FIGURE 19 AVERAGE INSTANTANEOUS IMPEDANCE ρ<sub>0</sub>D<sub>9</sub> AS A FUNCTION OF AXIAL STRESS FROM 50-kber EXPERIMENTS IN ALUMINUM

(Eqs. 100 and 101 of Section III.1) in the stress-particle velocity and the stress-volume planes. These integrations result in the compression paths, Hugoniot points, and release adiabats from Egs. 20 and 21. As expected, both of these figures show a significant unference between loading and unloading paths. This hysteretic effect contributes to energy losses and attenuation.

The measured Hugoniot elastic limits (HEL) for these aluminum alloys (4.3 kbar for 6061-T6 and 5.5 kbar for 2024-T8) are in fair agreement with the static yield strength from uniaxial stress experiments. According to either von Mises or tresca yield criteria (with a Poisson's ration of 0.32), the HEL should be 1.9 times the static yield strength, if strain rate effects are negligible.

The relation between HEL and yield strength is given in Section V.2 as Eq. 104.

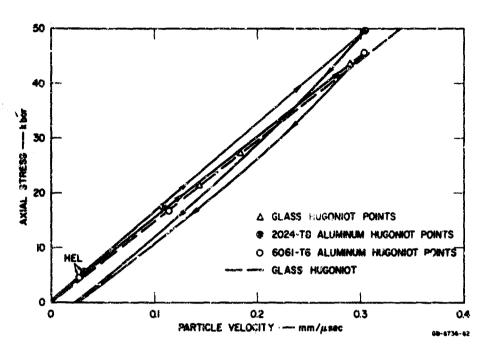


FIGURE 20 . STRESS-PARTICLE VELOCITY COMPRESSION AND RELEASE PATHS FROM 50-kbar EXPERIMENTS IN ALUMINUM

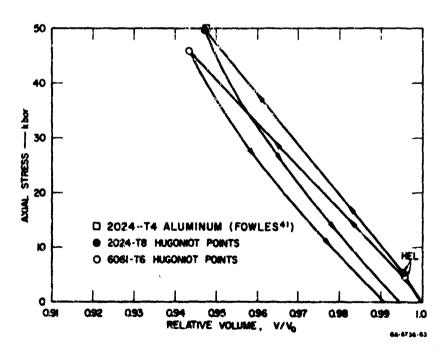


FIGURE 21 . STRESS-VOLUME COMPRESSION AND RELEASE PATHS FROM 50-kber EXPERIMENTS IN ALUMINUM

While the HEL's do not show a strain rate effect, the stress-time profiles of Figs. 16 and 17 show a nonsteady-state yielding zone (elasticplastic transition) that extends from the HEL to approximately 1.8 times the HEL for both aluminum alloys. The wave propagation in this yielding zone is dispersive with a time-independent, stress-dependent local disturbance velocity, a condition which generates a single distanceand time-independent compression path in the stress-particle velocity and stress-volume planes. The HEL is the stress at which dynamic yielding begins (corresponding to the proportional limit of tensile experiments) and may be measured by noting the maximum stress for which the elastic precursor is steady state. The HEL one obtains by this definition differs little from that obtained by other definitions based on the shape of the precursor profile. However, this definition provides a more rational basis for the selection of the HEL since the slope of the stress-strain curve will be a straight line up to this stress, but not beyond. The fact that yielding extends over a wide stress range for dynamic loading suggests that rate-dependent phenomena may play a role in the precursor formation. However, the HEL did not vary with propagation distance in these experiments, probably because of the relative thickness of the targets studied. The HEL would be expected to attenuate with propagation distance at very early times and over very short distances shape between the HEL and the main wave can be predicted by a rateindependent mechanism: the rounded stress-volume relation in the vicinity of the yield point.

Release adiabats measured with the above-described multiple gage technique were markedly different from those predicted by simple elastic-plastic theory. The elastic release modulus and the yield strength were both larger than predicted, and no well-defined yield point was observed during release.

Rarefaction velocities were obtained as a function of stress level from the low-stress experiments. This computation was possible because of the load impedance match between the gage material and the samples. The rarefaction velocity was obtained by observing the first reduction

in stress from a flat top in the Hugoniot experiments. The resulting rarefaction velocities are shown in Table 5 as a function of stress level. As expected these velocities are well above the zero stress sound velocities. Considerable judgment is involved in selecting the point of arrival of the rarefaction wave, and therefore it is not surprising that there is some scatter in the data.

# b. HE Experiments on Aluminum

The HEL was overdriven in all HE experiments on aluminum alloys. The Hugoniot data from the HE experiments were obtained using techniques described in Section III.2.b and the results are presented in Figs. 13 and 14 and Tables 3 and 4. These results are in good agreement with other Hugoniot data on these materials as shown in Figs. 13 and 14. To allow measurements of the stress histories during loading and unloading, three thicknesses of each alloy (approximately 3/8 inch, 1/2 inch, and 1 inch) were impacted by 1/8-inch-thick stress-free flyer plates traveling at about 3 mm/µsec (HE system E). The shot configurations are shown in Fig. 9 and the instrumentation is described in Section III.2.b. Profiles for each of these thicknesses obtained from the manganin C-7 transducers are shown later in the section (Figs. 24 through 26 and 30 through 32), in combination with the corresponding calculated stress histories. The 1-inchthick target experiments were designed to show peak stress attenuation; therefore these experiments do not yield Hugoniot data and do not appear in Tables 3 and 4.

# 3. Calculations for 2024-T8 Aluminum

Experimental stress records used for calculational comparisons of shock attenuation were obtained at two stress levels in 2024-T8 aluminum: 50 kber and 300 kber. The high stress data were obtained with a manganin transducer embedded in C-7 epoxy. The impedance mismatch between C-7 and aluminum is so severe that the resulting stress records could not be readily used with the analysis procedure described in Section III.1. The tests at 50 kber employed a manganin gage embedded in glass with an impedance very similar to that of the aluminum. Therefore, the latter data were used to develop the Bauschinger parameters. PUFF calculations

Table 5
ALUMINUM PEAK RAREFACTION VELOCITIES

Material	Initial Velocity* (mm/µsec)	Peak Stress (kbar)	Stress (kbar)	Peak Rarefaction ** Velocity (mm/µmec)
6061-Т6	6.2	45.6	45.6	7.3
60 <b>61-T6</b>		45.6	40.0	6.8
6061-T6		45.6	30.0	6.1
6061-T6		45.6	20.0	5.6
6061-T6		45.6	10.0	5.3
6061 <b>-</b> T6	6.2	17.0	17.0	6.8
6061-T6		17.0	10.0	6.2
6061-T6		17.0	5.0	5.6
2024-T8	6.5	49.6	49.6	7.7
2024-TS		4° 6	45.0	7.4
2024-T8		49.6	40.0	7.0
2024-T8		49.6	30.0	6.4
2024-T8		49.6	20.0	5.9
2024~T8		49.6	10.0	5.6
2024-T8	6.5	17.5	17.5	6.8
2024-78		17.5	10.0	6.1
2024-T8		17.5	5.0	5.4

Wave velocity of the compressional precursor at the zero stress level, (equal to the longitudinal sound velocity).

<sup>\*\*</sup> Wave velocity of lewing edge of rarefaction fan measured with respect to the material at that point.

were then made to compare with the experimental records at both stress levels. The stress gage data from four low-pressure experiments on 2024-T8 aluminum are shown in Fig. 16 on a common time base, with the impact time at zero. Sets of values of \$\rho\$ D were obtained from each pair of records through the use of Eq. 87. With measurements on all possible pairs of records, six curves of  $\rho$  D vs stress were obtained for the 2024-T8 aluminum. The composite  $\rho$  D-stress curves shown in Fig. 19 were the basis of a numerical integration of Eq. 101 to obtain the stress-volume relation shown in Fig. 22. To guide in the interpretation of the relation, a computed pressure-volume relation is also shown. The loading curve of Fig. 22 exhibits an elastic limit at 5 kbar, but the Rayleigh line passes between 10 kbar and the peak stress. Hence, there is a range of gradual yielding from 5 to 10 kbar. This effect is more evident in the plot of deviator stress, Fig. 7. (The deviatoric stress curves are taken from the difference between the stress and pressure curves of Fig. 22). The 5-kbar level is the magnitude of the elastic precursor. The 5 to 10 kbar yield region is associated with a gradually increasing stress between the precursor and the main wave. The unloading

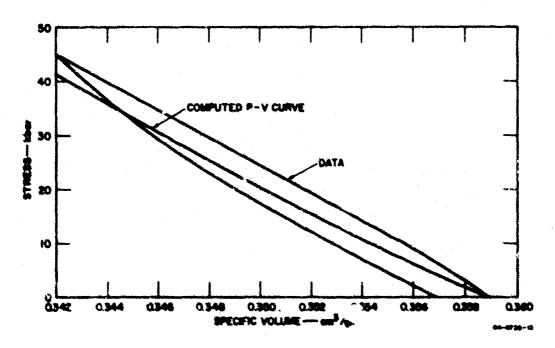


FIGURE 22. STRESS-VOLUME PATH REDUCED FROM DATA FOR 2024-TB ALUMINUM AND A COMPUTED PRESSURE-VOLUME RELATION

curve is smooth: the sharp unloading yield point predicted by the model of ideal plasticity is not present. The slope of the unloading curve at the peak stress is noticeably steeper than the loading curve at zero stress. This increased steepness is probably associated with the larger (10 to 20 percent) bulk modulus and larger shear modulus at the higher stress. Near the point of zero stress after unloading, the deviatoric stress and hence the yield strength appears to be two or three times as large as they were during the initial loading.

The ten unknowns introduced in Eqs. 60 through 65 for the Bauschinger model -- C, D, S,  $M_0$ ,  $M_1$ ,  $M_2$ ,  $M_3$ ,  $Y_1$ ,  $N_1$ , and  $N_2$  -- were determined by matching the experimental load-unload S-V curve with the analytical curves. This required a certain amount of frial and error and the resulting set of values is by no means unique. The form of the modulus variation appeared to represent the data very well: the fit to the unloading curve is shown in Fig. 7 for  $N_2 = 1.5$ . The computed values of the parameters are shown in Table 6.

Using the Bauschinger parameters from Table 6, a PUFF calculation was made to compare with the experimental stress records shown in Fig. 16. After several tries we obtained the set of computed records shown in Fig. 23. These records were obtained using the parameter values in parentheses in Table 6. The rarefaction velocity was too small so it appeared necessary to augment the unloading shear sodulus. In fact, there is still not enough attenuation in the computed records so that possibly even a higher  $M_3$  should have been used. The curvature  $N_1$  of the initial loading function was essentially eliminated by giving it a large value. With  $N_1 = 6000$ , the function has a sharp yield. The computed stress records, however, show about the right magunt of rounding at the precursor front because of the artificial viscosity.

The comparison of experimental and computed stress records in Fig. 23 exhibits the following features:

- The arrival times are satisfactorily predicted by the code computations
- e Peak stress is overestimated; some of the overestimate is caused by the overshooting which is characteristic of this code, some is caused by an inadequate attenuation of stress, and some by what appears to be too large values of C, D, and S.

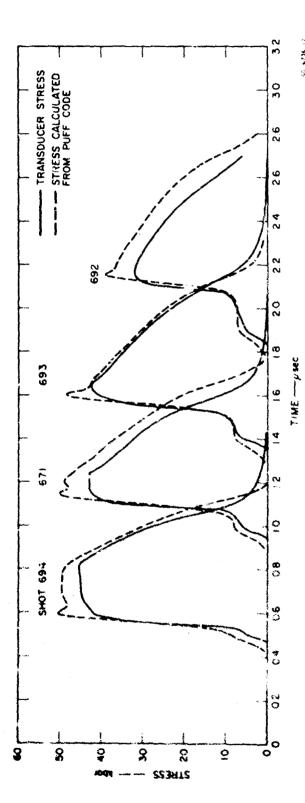


FIGURE 23. COMPARISON OF COMPUTED AND RECORDED STRESS HISTORIES FROM IMPACTS WITH 2024-TB ALUMINUM AT 50-kber

Table 6

PARAMETERS FOR THE BAUSCHINGER MODEL
FOR 2024-TE and 6061-T6 ALUMINUM

Parameter	2024-T8	6061-T6
EQSTC (dynes/cm <sup>2</sup> )	7.60 × 10 <sup>11</sup>	6.67 × 10 <sup>11</sup>
EQSTD (dynes/cm²)	1.50 × 10 <sup>13</sup>	(1.0 × 10 <sup>12</sup> )
M <sub>1</sub> (dynes/cm <sup>2</sup> )	3.0 × 10 <sup>11</sup>	2.67 × 10 <sup>11</sup>
N <sub>2</sub> (dynes/cm²)	7.63 × 10°	3.79 × 10 <sup>10</sup>
M <sub>3</sub> (dynes/cm²)	$4.45 \times 10^{11}$ $(4.82 \times 10^{11})$	4.67 × 10 <sup>11</sup>
N <sub>4</sub> (dynes/cm <sup>2</sup> )	2.18 × 10 <sup>10</sup>	3,79 x 10 <sup>10</sup>
Y <sub>1</sub> (dynes/cm <sup>3</sup> )	4.0 × 10 <sup>9</sup>	3,21 × 10 <sup>9</sup>
N <sub>1</sub>	6.0 (6000)	6.0 (6000)
иа	1.5	1.0

Note: Parenthetical values are the final values obtained by comparing the PUFF calculations with the experimental stress records.

 The unloading arrival is fairly well predicted, and the shape of the unloading curve is well matched.

Figures 24 through 26 show the experimental data from the 300 kbar shots and the corresponding computed records. The computed predictions are very close for the thin targets (Figs. 24 and 26) but the computed stress is 20 percent low for the thickest target. These calculations were made using the same Bauschinger parameters as for the low stress computations. Hence, it is not surprising that there is some disagreement in magnitude.

We conclude then from the comparison of all the results for 2024 that the form of the Bauschinger model is well-suited to represent 2024-T8 aluminum; however, some changes in the parameters are required to quantitatively predict stress amplitudes.

## 4. Calculations for 6061-T6 Aluminum

Impact tests used for calculational comparisons were conducted at 50 and 300 kbar, the high stress tests with a manganin C-7 gaze and the low stress tests with a manganin glass gage. The series at the low stress level was used in the analysis presented in Section III 1 to obtain the Bauschinger and equation-of-state parameters. Then PUFF calculations were made to compare with the experimental records at both stress levels.

The determination of Bauschinger parameters will be sketched briefly here: it follows the same procedure as that for the 2024-T8 aluminum. The series of experimental stress records is shown in Fig. 16. The plot of  $\rho$  D versus stress derived from these records is shown in Fig. 19. The resulting stress-volume plot and stress deviator plot are shown in Figs. 27 and 28. The computed Bauschinger parameters are listed in Table 6. The PUFF computations based on these parameters are shown in Figs. 29 through 32. In general the results of comparing the computed and experimental curves are similar to the results for 2024-T8 sluminum. There is qualitative agreement between prediction and experiment but attenuation predicted is not sufficient; hence, the unloading, or rarefection, velocity is higher than expected. The differences that

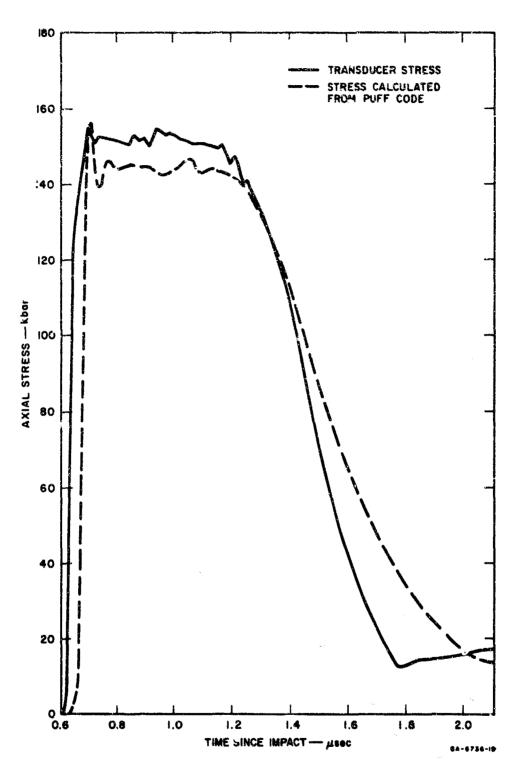


FIGURE 24. COMPARISON OF COMPUTED AND RECORDED STRESS HISTORIES FROM SHOT 13,532 IN 4.48-mm-THICK 2024-T8 ALUMINUM TARGET

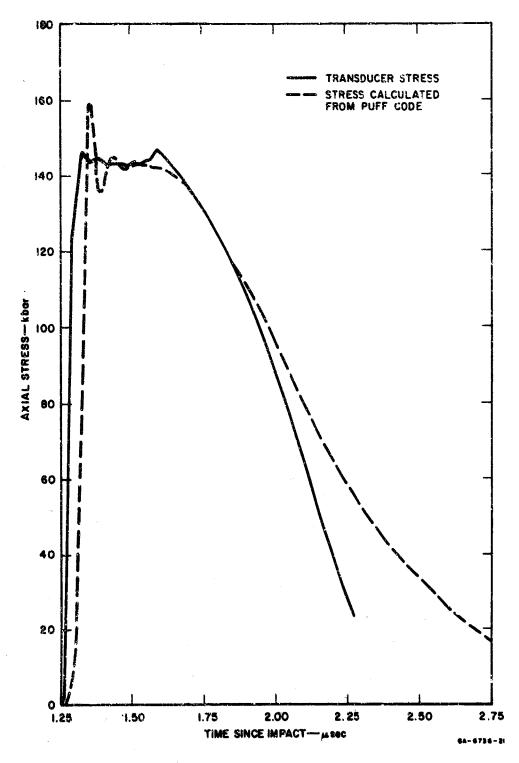


FIGURE 25. COMPARISON OF COMPUTED AND RECORDED STRESS HISTORIES FROM SHOT 13,532 IN 8.96-mm-THICK 2024-T8 ALUMINUM TARGET

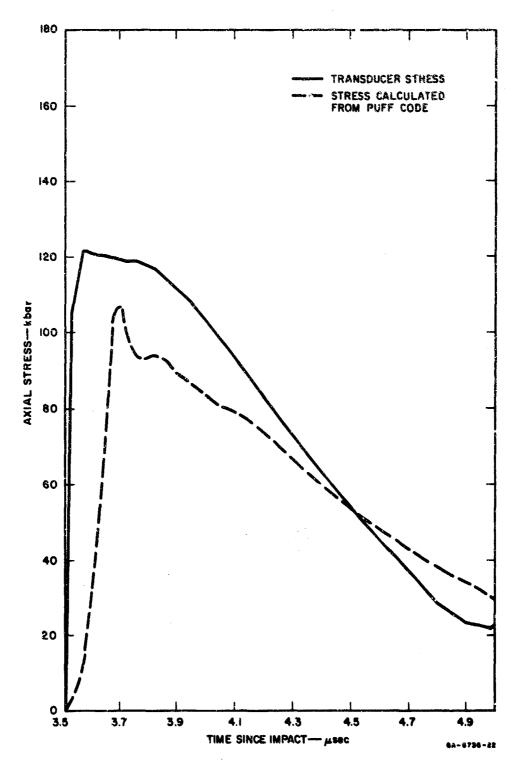


FIGURE 26. COMPARISON OF COMPUTED AND RECORDED STRESS HISTORIES FROM SHOT 13,529 IN 25.42-mm-THICK 2024-T8 ALUMINUM TARGET

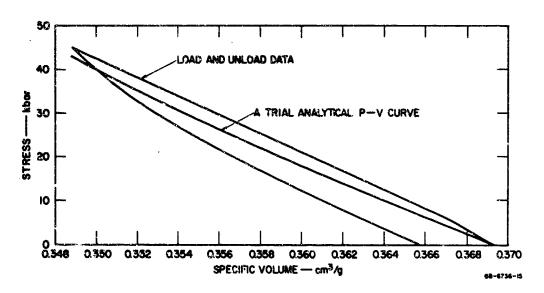


FIGURE 27 STRESS-VOLUME PATH REDUCED FROM DATA ON 6061-T6 ALUMINUM AND A COMPUTED PRESSURE-VOLUME RELATION

exist between the calculated and measured unloading portions of the waves indicate that different values of the Bauschinger parameters should permit a better quantitative match between computation and experiment.

## 5. Summary of Results

The results of experiments and theoretical calculations with 2024-T8 and 6061-T6 aluminum fall into three main categories:

- Bauschinger effect: pronounced Bauschinger effect in both alloys governed the speed of rarefaction waves, the shape of the unloading portion of stress waves, and the stress attenuation rate. The apparent shear modulus was 60 to 70 percent larger for unloading from 50 kbar than for initial loading. No unloading yield point was observed; rather a gradual transition from elastic to plastic behavior during unloading, with a continuously decreasing shear modulus, was observed.
- No significant stress relaxation effects were observed in the dynamic experiments with either alloy for the specimen thicknesses (1/8 inch and larger) studied; this result is in accord with the experimental observation that little difference was observed between the static and dynamic yield strengths for plane strain. This result does not preclude the existence of stress relaxation effects at very early times.

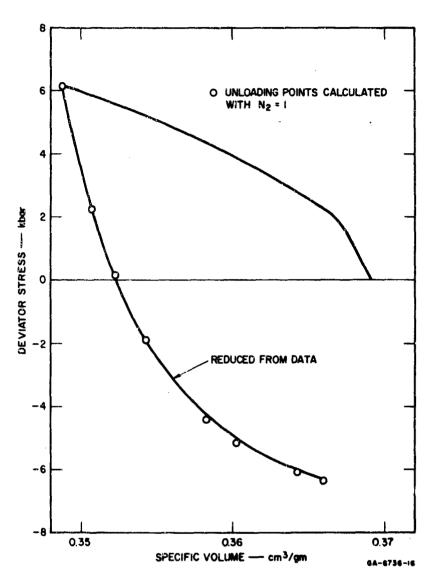


FIGURE 28. RELATIONSHI? BETWEEN DEVIATORIC STRESS
AND SPECIFIC VOLUME OBTAINED FROM 50-kbsr
TESTS IN 6061-T6 ALUMINUM

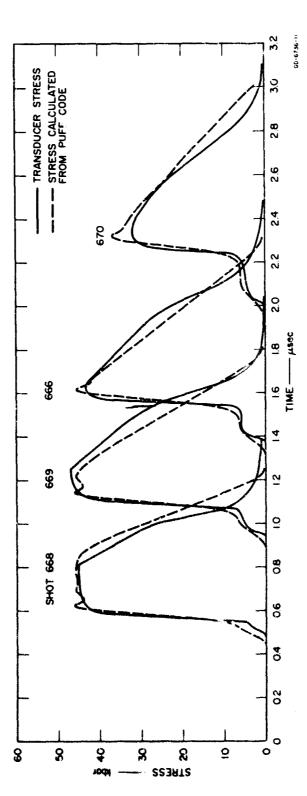


FIGURE 28, COMPARISON OF COMPUTED AND RECORDED STRESS HISTORIES FROM IMPACTS WITH 8061-T6 ALUMINUM AT 50-kbsr

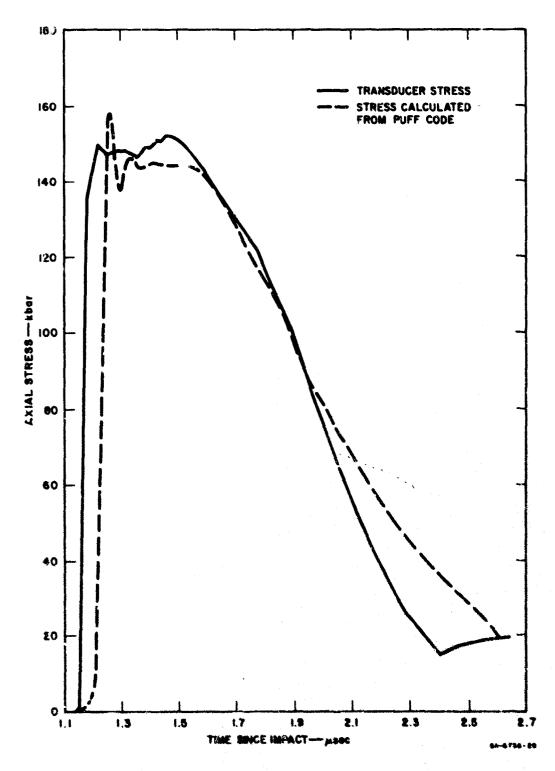


FIGURE 30. COMPARISON OF COMPUTED AND RECORDED STRESS HISTORIES FROM SHOT 13,550 IN 8.21-mm-THICK 8061-T6 ALUMINUM TARGET

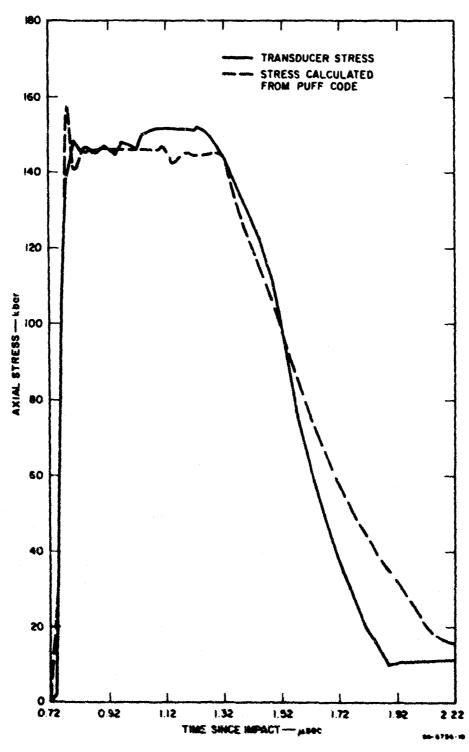


FIGURE 31. COMPARISON OF COMPUTED AND RECORDED STRESS HISTORIES FROM SHOT 13,569 IN 4.86-mm-THICK 6061-T8 ALUMINUM TARGET

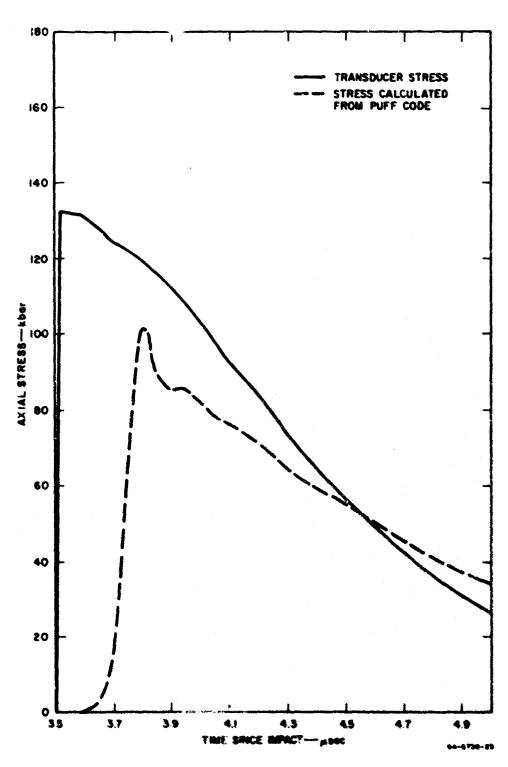


FIGURE 32. COMPARISON OF COMPUTED AND RECORDED STRESS HISTORIES FROM SHOT 13,530 IN 25-mm-THICK 6081-T8 ALUMINUM TARGET

• A large elastic-plastic transition region in the stressvolume plane was observed in the vicinity of the HEL for
both alloys. This convex-upward region if negative
curvature leads to a gradually rising nonsteady-state
stress profile between the precursor and main wave.

Errors of 5 or 10 percent may result in calculated Hugoniot
points if this region is ignored and a simple one- or twowave structure assumed instead. Fair correlation was
obtained between the stress records and the stress histories
calculated on the basis of the Bauschinger model in the SRI
PUFF 2 code. The model appears to represent both alloys
qualitatively, but the numerical values of the model
parameters for each metal require further investigation.

### SECTION V

### STUDY OF TITANIUM

### 1. Introduction

The preliminary investigation of titanium alloys was directed toward establishing Hugoniots in the low stress range, measuring shock attenuation, examining strain rate effects, looking for phase transitions, and observing the stress wave appearance.

For structural purposes, titanium is usually alloyed with other materials to improve its mechanical properties. Pure titanium is in the alpha (hcp) crystal phase at room temperature. Eut with the addition of certain alloying metals, the high-temperature beta (bcc) phase can be retained at room temperature.

These alloys were selected: type 50A, commercially pure (99.2 percent) titanium, that is, all alpha phase; Ti-6Al-4V (6 percent aluminum, 4 percent vanadium), an alloy with mixed alpha and beta phases; and Ti-13Cr-11V-3Al (73 percent titanium, 13 percent vanadium, 11 percent chromium, 3 percent aluminum), an alloy that is all beta phase. The Ti-6Al-4V alloy was selected because it is a material of practical importance that has been studied at intermediate strain rates, where it exhibited strain rate sensitivity (Ref. 8). The 50A titanium and Ti-13Cr-11V-3Al were chosen to study independently the behavior of each of the constitutent phases of the Ti-6Al-4V alloy.

The titanium alloys were supplied by Titanium Metals Corporation of America, Los Angeles, in thicknesses of 1/16 inch to 1-1/8 inches. It did not appear to be economical nor possible within the time requirements to form all sheets of an alloy from a single billet. Therefore, the manufacturer selected sheets from billets with a nearly identical chemical composition. All sheets were supplied fully annealed, and no tempering or work hardening was performed at SRI on the cheets.

A table of quasi-static mechanical properties of these alloys is included here for reference (Table 7). Some of the information was obtained from our quasi-static and acoustic measurements listed in Appendix II, some from Molchanova (Ref. 42), from Titanium Metals Bulletins

Table 7
PROPERTIES OF TESTED TITANIUM ALLOYS

Alioy	Density (g/cm <sup>3</sup> )	Young's Modulus (kbar)	Yield Strength (kbar)	Poisson's Ratio	Phase <sup>a</sup>
Type 50A	4.507	1070	3.5	6.328	alpha
Ti-6Al-4V	4.424	1210	9.5	υ.314	alpha and beta
Ti-13V-11Cr-3A1	4.839	1040	9.3	0.032	beta

Alpha phase is hexagonal, close packed; beta is body-centered cubic. The percentage of alpha and bet in Ti-6Al-4V depends on the heat treatment. Both Ti-4Al-4V as 1-13Cr-11V-3Al are heat-treatable.

(Ref. 42, 44), and from Harman (Ref. 45). The Young's modulus varies from 1000 to 1200 kbar for all the heat-treatable alloys, depending on the heat treatment. It is lowest, of course, for annealed alloys (and for pure alpha, which is not sensitive to heat treatment), higher for solution treated alloys, and highest for aged and solution treated. Hence, particularly at low stresses, one would not expect a unique Hugoniot for a heat-treatable alloy. The Hugoniot for pure titanium is given by McQueen and Marsh (Ref. 46).

$$U_S = 4.779 \times 10^5 + 1.089 U$$
 (102)

where  $\mathbf{U_S}$  and  $\mathbf{U}$  are shock and particle velocity in centimeter/second. The corresponding PUFF-form for the Hugoniot is

$$P = 1.028 \times 10^{12} \mu + 1.21 \times 10^{12} \mu^{2} + 2.06 \times 10^{11} \mu^{3}$$
 (103)

where is pressure in dyne/cm2.

## 2. Dynamic Experiments

Both gas gun and HE impact experiments were conducted on the three chosen titanium alloys. The experimental arrangement for the gas gun tests was similar to that used for the aluminum impacts (Fig. 8), except that titanium flyer plates were bonded directly to the aluminum projectile and manganin in Hi-D glass transducers were used. Configuration B (Fig. 9) with manganin in C-7 transducers and optical instrumentation were used for the HE experiments. The Hugoniot data obtained from these tests are summarized in Figs. 33 through 36 and Table 8.

The present data for type 50A titanium agrees closely with data from studies of Walsh et al. (Ref. 47) and McQueen and Marsh (Ref. 46). They used a gap technique that is not capable of resolving multiple shock waves associated with phase transformations. In spite of the phase transformation (described later) beginning around 50 kbar, there is good agreement between our data and theirs. This indicates that the dynamic phase transition is associated with only a small overall density change and has little effect on shock velocity.

The Hugoniot points above 100 kbar were obtained with the HE system. In all HE experiments the elastic wave was overdriven; therefore, a steady-state, single-wave analysis procedure (the impedance match method) was used to reduce the data.

In the gas gun experiments, the maximum stresses were below 100 kbar. The precursor, a second wave associated with a phase change in the case of pure titanium, and a gradually increasing stress between waves were in evidence in the stress transducer records. Two of these records are shown in Fig. 37. The upper figure shows the wave aused by the phase transformation. These complexities in the wave front necessitated the use of the integration procedure of Section III.1 to determine the stress-volume and stress-particle velocity relations. The results of these integrations are shown in Figs. 35 and 36.

The static yield strengths and Hugoniot elastic limits from the gas gun experiments on the three alloys are listed in Table 9. (Neglecting strain-rate effects, the HEL and static yield strengths of an isotropic, \* See Section 111.2.c

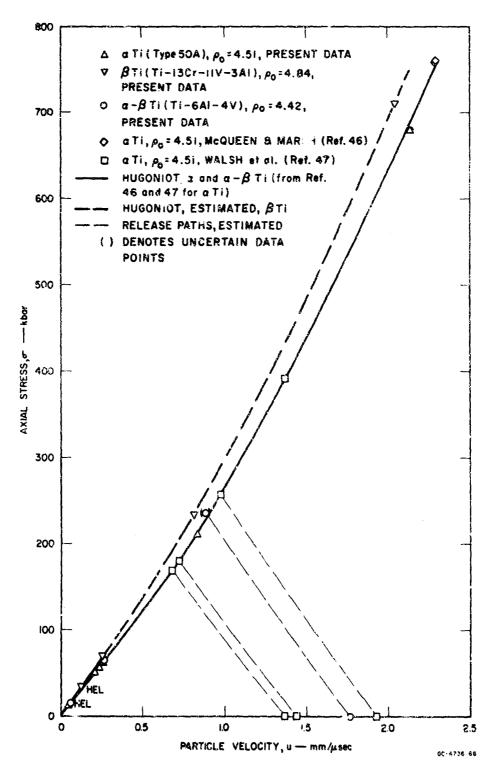


FIGURE 33 STRESS-PARTICLE VELOCITY HUGONICT AT SE DATA FOR TITANIUM ALLOYS

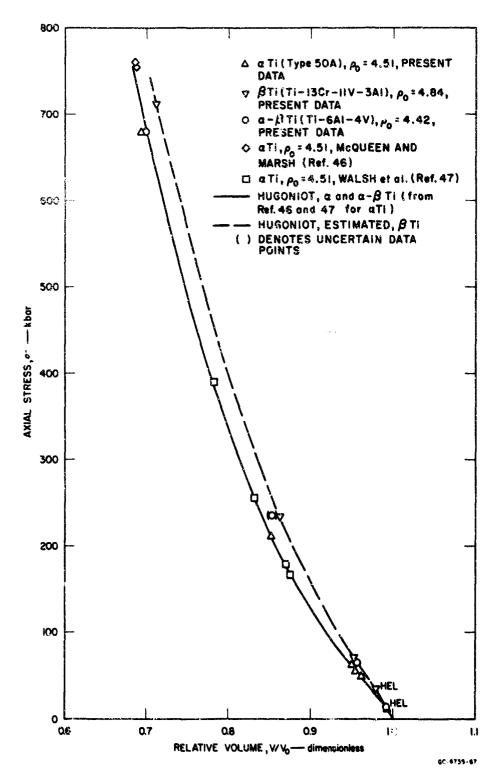


FIGURE 34 STRESS-VOLUME HUGONIOT DATA FOR TITANIUM ALLOYS

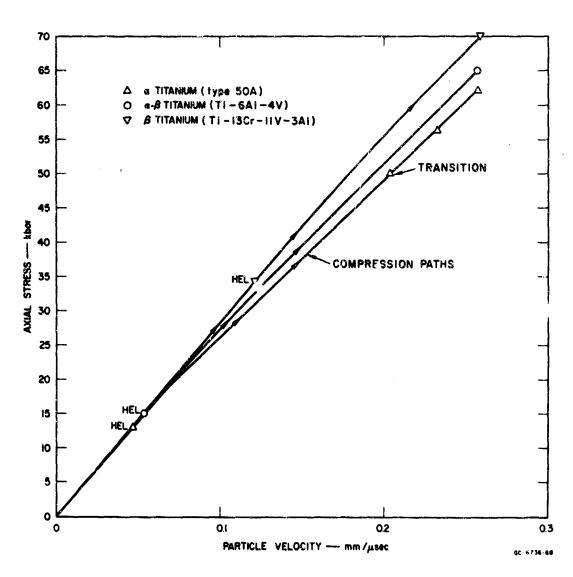


FIGURE 35 STRESS-PARTICLE VELOCITY HUGONIOT DATA AT LOW STRESSES FOR TITANIUM ALLOYS

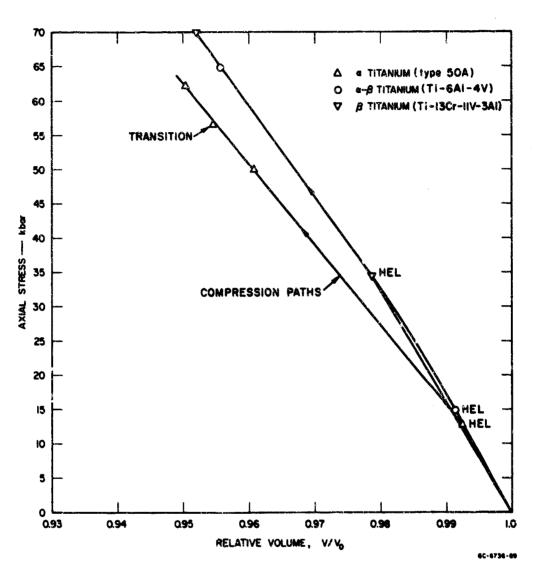


FIGURE 36 STRESS-VOLUME HUGONIOT DATA AT LOW STRESSES FOR TITANIUM ALLOYS

Table 8 HUGONIOT DATA FOR THREE TITANIUM ALLOYS

 $\alpha$  Titanium, Type 50A,  $\rho_0 = 4.51 \text{ g/cm}^3$ 

<del></del>				Final Hugon	iot State	
Shot No.	Driver System	Specimen Thickness (mm)	Axial Stress C (kbar)	Particle Velocity u (mm/#sec	Specific Volume V (cm <sup>3</sup> /g)	Shock Velocity U (mm/µmec)
G 589	Gas gun; projectile velocity = 0.466 mm/µsec	9.93	56.5	0.233	0.211	
G <b>604</b>	Gas gun; projectile velocity = 0.515 mm/µmec	9.95	62.2	0.258	0.210	
73,526 <sup>‡</sup>	HE system D; driver free surface velocity = 1.56 mm/Mec	5.08	211	0.835	0.189	5.63
13,527	HE system B; driver free surface velocity = 4.85 mm/#mec	4.83	679	2、14	0.154	7.01
			<b>8</b> 1	itanium, **	3Cr-11V-3A1,	0 <sub>0</sub> = 4.84 g/c
G 819**	Ges gun; projectile velocity = 0.519 mm//mec	6.97	70	0.259	0.197	
13,524 ††	HE system Di <sup>5</sup> driver free surface velocity = 2.04 um/usec	6.76	234	0.815	0.178	5.94
13,527	NE system B; driver free surface velocity = 4.85 nm/ heec	6.54	712	2.05	0.147	7.13
		<u>_</u>	α -	β Titanium, †††	1-6A1-4V, p	4.42 g/cm <sup>3</sup>
G 791 **	Ges gun shots, everage	19.5	65.0 <sup>95</sup>	0.258 <sup>†§§</sup>	0.216	
G 818 **	Projectile velocity = 0.516 99 ± 0.001 mm/phoc	5.72	•	V1130	J. 2.2	
13,545	HE system D; <sup>§</sup> driver free surface velocity = 2.10 mm/µme	c 5.52	(237)***	(0.885)***	(0.193)***	(5.08)***
13,527	ME system B; driver free surface velocity = 4.85 mm//ase	c 4.84	712	2.05	0.147	7.13

The following intermediate states were observed on Shot G589: an HKL of  $\sigma$  = 12.8 kber, u = 0.0404 mm/gbec, V = 0.219 cm<sup>3</sup>/g; and a phase transformation at  $\sigma$  = 50.0 kber, u = 0.204 mm/gbec, V = 0.213 cm<sup>3</sup>/g.

<sup>†</sup> Taken to be one-half the projectile velocity.

<sup>\*</sup> In Shot 13,523 a C-7/manganin stress transducer on the Ti specimen recorded a final stress ... 82 hber. The on-Rugoniot perticle velocity associated with 82 kber is u = 1.45 mm/phec. This O -u state represents an experimental determination of a point on the release adiabet on Ti shocked to 211 kber.

See Table 1 for description of HE systems.

<sup>&</sup>lt;sup>40</sup> On Shot G 819 the MEL was measured to be  $\sigma$  = 34.4 kbar, u = 0.1215 mm/place, V = 0.202 cm<sup>2</sup>/g.

 $<sup>^{\</sup>dagger\dagger}$  Qm Shot 13,524 the specimes free-surface velocity was observed to be 1.9  $\pm$  0.3 mm//mec. The large experimental uncertainty is caused by a large shock tilt.

On Shote G 781 and G 818 the averago MEL was determined to be G=18.0 kber, u=0.0538 mm/ $\mu$ mec, V=0.224 cm<sup>3</sup>/g.

fo Averaged date from Shots G 791 and G 818.

Large experimental uncertainty due to poor transit time mirror cutoffs on streak camera record.

Table 9

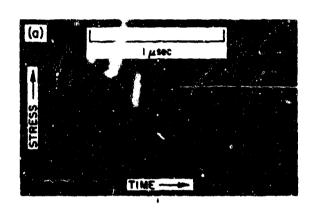
COMPARISON OF STATIC AND DYNAMIC YIELD STRENGTH IN TITANIUM

Alloy	Static Yield (kbar)	Hugoniot Elastic Limit (kbar)	HEL predicted from static yield and Poisson's ratio (kbar)
Type 50A	3.5	12.8	6.85
T1-6Al-4V	9.5	15	17.5
Ti-br-11V-3Al	9.3	34.4	18.4

homogeneous material obeying a von Mises or Tresca-type yield criterion are related by

$$G_{HEL} = \frac{1 - \nu}{1 - 2 \nu} Y$$
 (104)

This equation was used to calculate the values in the fourth column of the table.) The HEL's for the type 50A tit nium and Ti-13Cr-11V-3Al alloys are nearly double those calculated from the static data (column 4 of Table 9). By constrast, the observed HEL for Ti-6Al-4V is lower than that calculated from static data. In the stress records the Ti-6Al-4V alloy exhibited a wide yielding zone between the precursor and main wave. extending from the HEL to 2.3 times the HEL (an example of such a range is shown in Fig. 37b). The other alloys showed a smaller yielding zone (in type 50A, the yielding zone extended only to 1.3 times the HEL). The extensive yielding zone in Ti-6Al-4V is caused by a significantly rounded yield "point" in the stress-strain curve. This rounded yield point also helps to explain the occurrence of a low HEL. The experiment lly measured HEL actually corresponds to the "proportional limit," the stress at which the st ess-strain curve first deviates from a linear relation. Normally, the proportional limit is somewhat smaller than the so-called yield strength (0.2% offset definition used here). For a well-rounded yield point, the proportional limit could be much lower than the 0.2% offset yield point.



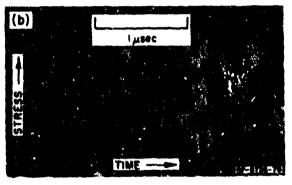


FIGURE 37 MANGANIN STRESS RECORDS FROM TEST G589 ON TYPE 50A TITANIUM AND TEST G791 ON TI-6A1-4V

The disparity in dynamic yield behavior of the three titanium alloys indicates that low strain rate yield data are not adequate to predict dynamic yielding in these metals. The large increase in yield strength for alpha and beta titanium at high strain rates is some justification for the inclusion of a rate-dependent yield criterion in wave propagation calculations. However, attenuation of the elastic precursor (a definite sign of strain-rate effects) was not observed in any of the titanium experiments over propagation distances ranging from 5 mm to 9.5 mm. This does not imply that stress relaxation does not occur over distances less than 5 mm. The work of Isbell and Froula (Ref. 48) Gm very thin targets indicated that although stress relaxation was important over short distances, no further relaxation occurred after propagation distances of 7 mm; this is consistent with our results.

As mentioned above, indication of a dynamic phase transition was observed in type 50A titanium (see Fig. 37a). Shock loading to a peak stress of 56.5 kbar caused the plastic wave to split into two waves, which is the effect produced by transformation times that are comparable to or less than the wave transit time. The separation of these waves was very small even after transit through a 3/8-inch-thick sample, indicating that the overall volume change associated with this transition or pactial transition is small (approximately 0.2 percent). The transition stress was 50 kbar and the transition was overdriven in subsequent experiments at a stress of 62 kbar. Our observation of such a transition was subsequently correborated by Isbell and Froula (Ref. 48). No such transition wave was observed in either Ti-6Al-4V or Ti-13Cr-11V-3Al titanium; however, any such transition is expected to occur at different pressures in these alloys, and the ease with which the transition wave can be overdriven in 50A titanium suggests that it may take considerable investigation to discover a like transition in the other two alloys.

Rarefaction velocities were obtained as a function of stress level from the low stress experiments. These velocities are shown in Table 10 as a function of stress level. These rarefaction velocities appear to coincide with the shock velocity of the HEL (except in the case of the all

Table 10

RAREFACTION AND SHOCK VELOCITIES IN TITANIUM ALLOYS

			Wave	Velocities	
Shot No.	Material	Pesk Stress (kbsr)	To HEL	To Peak (man/µsec)	Initial Release from Peak
G 604	Type 50A	62.0	6.1	5.05	6.1
G 791	T1-6A1-4V	65.0	6.3	5.20	6.3
G 818	T1-6A1-4V	65.0	6.3		6.3
G 819	Ti-13Cr-11V-3A1	70.0	5.8		6.2

beta alloy). Such a coincidence of rarefaction and initial shock velocity is actually predicted by the model of perfect plasticity. Hence, this simple plastic model might be sufficiently accurate for titanium alloys: the more complex Bauschinger model would not be required. A determination of the appropriate model should, however, be made from a comparison of computed and experimental stress records.

## 3. Initial and Terminal Observations of Shocked Titanium Specimens

Initially, the 50A titanium ("ss-received") had slightly elongated grains of about 30 \$\mu\$ diameter. The average Vickers hardness (500 gram load) was 142. Based on the relative intensity of X-ray reflection from flat speciments, one may conclude that there is some preferred orientation of the grains. After shocking to 56 kbar, the Vickers hardness of one specimen had increased to 200. Microstructural changes are clearly evident in this specimen. Nearly all the grains appear heavily marked with parallel sets of lamellae which, for the most part, completely traverse a grain (see Fig. 33). However, some grains show additional markings that do not appear to be twins but which more closely resemble a transformation structure. X-ray diffraction examination of the shock-loaded sample disclosed no evidence of new phases; so any transformation taking place during the shock must have reverted to alpha on release of pressure.

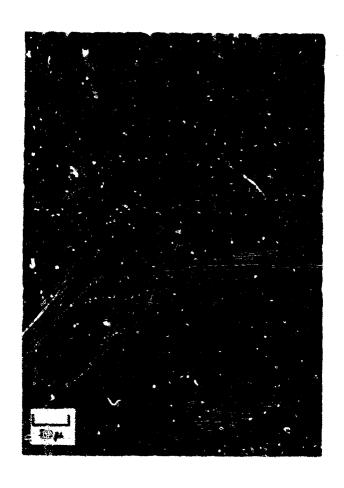


FIGURE 38 PHOTONECROGRAPH OF TYPE 50A
TITANIUM SAMPLE AFTER BEING
SHOCKED TO 56 kbm

Deribas et al. (Ref. 50) have described experiments in which titanium was shock hardened to 400 Vickers. They report a microstructure characterized by "immense distortion," similar to that observed in iron shocked to pressures in excess of lab kbar. They do not specify the pressure used to obtain this result; from thei: limited description of the experiments, we can estimate that pressures in the range of 150 to 400 kbar were used. While positive identification of the high-pressure phase is not possible, it is presumed to be the well-known omega (hexagonal) phase known to exist for some titanium and zirconium alloys. According to Jamieson (Ref. 49), the volume change for the alpha-omega transition in pure titanium is about 2 percent at 1 atm. He reported that the omega phase of titanium can be retained at 1 atm, but that it

reverts to the alpha structure after prolonged heating at 110°C. Deribas et al. (Ref. 50) did not report the presence of omega phase in their hardened specimens, and it appears likely that the residual temperature of the shocked litanium after release of pressure was too high to permit survival of omega in their experiments.

Our present interpretation of our experimental results is that a partial transformation of alpha titanium to the omega phase took place at a pressure of 50 kbar. (The pressure required for the static transition is not known). A volume change at the transition of at most 0.2 percent may be inferred from the transducer records. If one assumes that the volume change for the alpha-omega transition at 50 kbar is roughly the same as the 1-atm volume difference measured by Jamieson, it would appear that about 10 percent of the alpha titanium transformed in this experiment. This result is consistent with both the present metallographic results and with the increase in hardness observed. One may infer from the results of Deribas et al. (Ref. 50) that a large pressure excess is required to drive the transformation to completion in the microseconds or submicroseconds available in shock experimen's. Titanium thus appears to behave in a manner similar to that of iron (Ref. 51). Although the transformation of iron to the epsilon phase has been inferred to begin at shock pressures as low as 120 kbar, complete transformation requires a shock pressure of about 160 kbar. In the case of pure iron, the high pressure phase is not retained on release of pressure. The great hardening effect observed in shock loaded iron has been attributed to the dislocation structure produced by the transformation of epsilon to alphairon on release of pressure. Thus, alpha titanium appears to undergo a phase transformation initiated by a shock stress of 50 kbar and completed in the range of 150 to 400 kbar. The extent of transformation depends on the stress level reached. During unloading in titanium, the high pressure phase (probably omega) is retransformed to alpha with a considerably higher hardness than the original material.

The microstructure of the shocked slpha-beta alloy (Ti-6Al-4V) was indistinguishable from the unshocked material. The virgin material had a Vickers hardness of 300 (500 gram load). The shock loaded specimen had a very marginal increase in hardness to Vickers 318.

The microstructure of the shocked beta alloy (Ti-13V-11Cr-3Al) was distinguished by the presence of parallel sets of lamellae (see Fig. 39) presumably twins, which closely resemble the twi: formed by shock loading iron, molybdenum, tungsten, chromium, and other body-centered cubic metals and alloys below any transition pressure. The Vickers hardness of the unshocked material was 300 (500 gram load); the shocked specimen had the same hardness. It appears then that no transformations occurred in Ti-6Al-4V or Ti-13Cr-11V-3Al during the passage of shock waves with stresses up to 70 kbar.

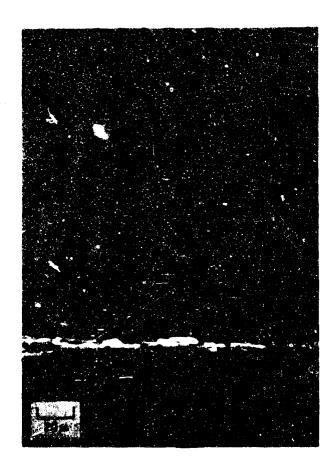


FIGURE 39 PHOTOMICROGRAPH OF TI-13V-11Cr-3A1 TITANIUM SPECIMEN AFTER BE:NG SHOCKED TO 70 kbar

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#### SECTION VI

### STUDY OF WOVEN QUARTZ PHENOLIC

# 1. Introduction

Preliminary experimental and calculational investigations have been performed to examine shock wave propagation properties between about 10 and 200 kbar for a three-dimensional (3-D) weave quartz phenolic material. The purposes of these studies were to obtain preliminary Hugoniot-type information for this material, to observe the character of stress waves that a highly anisotropic, heterogeneous material will support, and to determine how well present wave propagation theory, as incorporated in the SRI PUFF 2 code, is able to simulate stress wave profiles in an anisotropic, heterogeneous material.

### 2. Material Description

The specimen material, supplied by the Air Force Weapons Laboratory, was a composite consisting of an orthogonal 3-D weave of quartz filament potted in a phenolic resin matrix. Table II lists descriptive data supplied by AFWL with the specimens. The orthogonal weave of the quartz filaments is most easily visualized in terms of a Cartesian coordinate system with the Z axis in the direction of the axis of the right cylindrical specimen.

The weave was nearly orthogonal because all filaments lay approximately in the X. Y, or Z direction. In the Z direction the minimum were grouped together into well-contained, apparently straight bundles of square cross section with a 0.025 inch average width. These axial bundles intersected the X-Y planes (cylinder faces) in a square array with a 0.1 inch average distance center to center. According to Table 11 there were 0.58 x 10<sup>6</sup> axial filaments per square inch intersecting the X-Y planes. The filaments in the X and Y directions were also arranged into bundles, but of apparently rectangular cross section. The width of a lateral bundle was approximately

A true Hugoniot point refers to an equilibrium state achieved through a steady-state shock front. In the woven material, the wave front is nonuniform and not steady state. Equilibrium between the filaments and resin requires a relatively long time and is therefore probably not established during the recording time of the experiment. The Hugoniot-type data derived from the experiments were obtained with traditional impedance match techniques, neglecting the fact that many of the conditions for valid Hugoniot data were not met.

Table 11

DETAILED MATERIAL DESCRIPTION OF 3-D OUARTZ PHENOLIC \*

Property	Measurement
Average material density	1.69 g/cm <sup>3</sup>
Average acoustic velocity in axial direction	0.350 cm/µsec @ 1.0 MHz
Resin matrix	SC-1008 Phenolic
Resin content	53.7% by weight
Average diameter of filaments	0.00034 inch
Number of filaments in lateral (X) direction	2.30 x 10 <sup>8</sup> filaments/inch <sup>2</sup>
Number of filaments in lateral (Y) direction	2.30 x 10 <sup>8</sup> filaments/inch <sup>2</sup>
Number of filaments in axial (Z) direction	0.58 x 10 <sup>6</sup> filaments/inch <sup>2</sup>
Number of filaments per axial bundle	1920
Bundle size and spacing in the Z direction	0.025 on 0.1 inch centers

<sup>\*</sup> This information was supplied to SRI by AFWL.

0.075 inch and the thickness was about 0.025 inch. These lateral bundles were woven in a 2-D basket weave with, for example, a given X bundle passing over 1 to 8 Y bundles and under a similar number. The weave was not perfect but had an average or macroscopic rotational symmetry of 90° about the Z axis. The separation in the Z direction between parallel lateral bundles was about 0.05 inch center to center.

Acoustic measurements were made to determine the longitudinal velocity in the thickness direction. The velocity associated with the initial arrival averaged 4.51 mm/ $\mu$ sec. This velocity corresponds with a longitudinal molulus ( $\lambda + 2\mu$  or K +  $4/3\mu$ ) of  $3.4 \times 10^{11}$  dynes/cm<sup>2</sup>.

# 3. Gas Gun and HE Experiments

The compression states measured in five gas gun and HE experiments performed in the 3-D weave quartz phenolic study material ar summarized in Table 12 and Figs. 40 and 41; the results are discussed below. It must be pointed out that the nature of the quartz phenolic specimens, as well as direct experimental observations indicated that the flow was not steady-state. In this case the compressive "Hugoniot" states reported in Table 12 and Figs. 40 and 11 are characteristic of uniaxial flow shock wave experiments on 2 to 4-mm-thick specimens, but caution should be employed when applying them to other experimental conditions. For the purpose of describing gross average macroscopic response of material of a given thickness, we have neglected the deviations from steady-state flow.

Three gas gun shots were fired on the U-D quartz phenolic discs as received. Two discs were impacted by 1/2-inch thick 2024 aluminum projectiles to obtain Hugoniot data (Shots 13,421 and 13,423) and one by a 0.0186-inch-thick annealed aluminum flyer plate (backed by 2 lb/cu ft polyurethane foam) to obtain shock attenuation information (Shot 13,422).

<sup>\*</sup> These experiments were performed under Contract F29601-67-C-0073.

"HUGGHIOT" DATA FOR 3-D QUARTZ P-EMOLIC Table 12

3 bot System 13,423 Out 13,421	Flyer relocity (ma/wsec) 0.268			Dynamic		Pinal (Mugoniot) state	) state	-	Pinal state	Final state in C-7 (a release state)	case state)
	(m/usec) 0.268	thickness	density	Elastic Limit, Shock Velocity	Shock velocity,	Particle velocity,		Specific volume,	Axial stress,	Particle Telocity,	#13 e t1
<del></del>	0.248	ĵ	(g/cm²	(mm/~sec)	( <b>=</b> /256C	(BB/25C)	(kbar)	(8/,E)	(kber)	(mm/naec)	(2005/)
13,421		** 'd	1.67	·	3.02	102.1	10.1	0,560	5.7	0.17	0.89
	0.833	:	1.67	(1.17)	3,57	0.620	37	0.494	31	0.71	94.6
13,422	0.810	3	1.68	(4.52)	3.72	•	:	:	6.9	0.12	, ,
13,540 E(E)	3.08	4.57	79.		5.37	2.07	186	0.370 (0.354)**	(145)	(2.14)**	0.04

The dynamic electic limit was not well resolved and was . or used in calculating the final state. No precursor was observed in the ME experiments,

The release state was measured by a sabganin's stress transducer. Particle velocity was calculated from the C-7 Hugoniot of Keough (1964). In the gus experiments there ever two gages, one with the manginal element 0.015 inch back from the surface, the other 0.056 inch back. Data from the deeper element are scalified by the longer travel time in C-7 and are therefore listed in parentheses.

Projectile velocity for gwn system, flyer velocity or driver free-surface velocity for HE systems.

Also time for 10 to 90% of the peak atress, experimental uncertainty about 2.7.

Shocks front attendanted, Mugnetut state not calculable. State at impact was probably similar to that for Shot 13,421. Uncertain data. There was a large variation in sho, a arrival time.

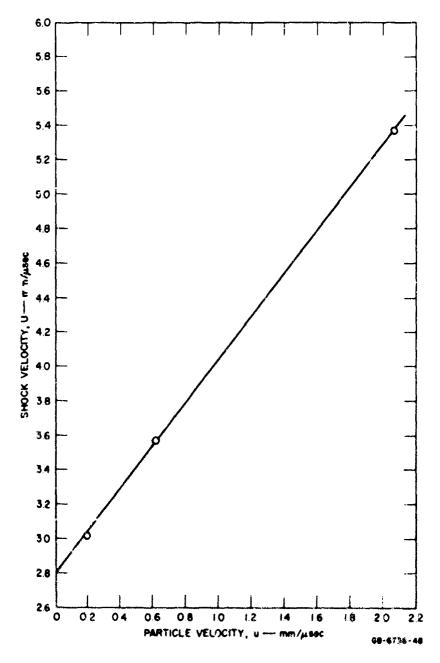


FIGURE 40 SHOCK VELOCITY-PARTICLE VELOCITY HUGONICT DATA FOR 3-D WEAVE QUARTZ PHENOLIC

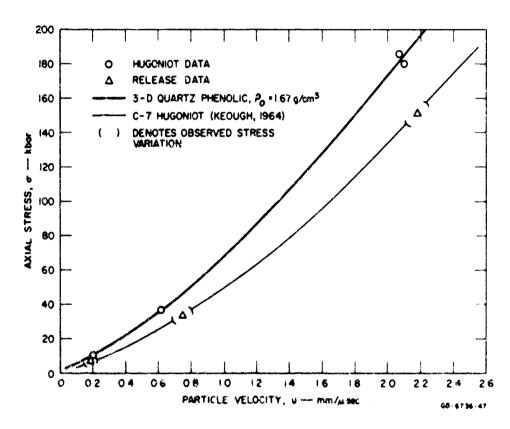


FIGURE 41 STRESS-PARTICLE VELOCITY HUGONIOT DATA FOR 3-D WEAVE QUARTZ PHENOLIC

Instrumentation on these shots consisted of two 10-chm manganin wire gages in C-7 epoxy. (High resistance 10-ohm gages are used in low stress experiments to improve the signal-to-noise ratio.) In one gage the mangan in element was positioned 0.015 inch back thio the C-i from the speciman - C-7 interface; in the other 1t was 0.080 inch back (the latter distance being about the same as the width of the lateral quartz fibers in the target). In all three gun shots the gage in which the element was set back 0.080 inch gave a shorter rise time and a lower stress for the main wave than the nearer gage. These changes in the wave appearance may be caused by convergence of the uneven wave front in the quartz phenolic into a planar wavefront in C7, damping of oscillations in the C-7, and tendency of the wave to shock up in the C-7. Because of the recorded evidence of nonsteady wave propagation in the C-7, the gage with the element nearer the interface was used to determine the reported loading state in the specimens. Those states were calculated by the impedance match method using measurements of the specimen wave velocities and the projectile velocities, using the known Hugoniat for the 2024-Al projectile.

Figure 42 shows a post-shot photograph of the recovered specimen and projectile head for Shot 13,421. The lateral quartz filament bundles appear as wide, limit traces in the photographs of both target and projectile, and form a nearly orthogonal grid. The small, dark squares represent the axial landles. These axial bundles left a deeper imprint on the aluminum projectile. The noticeable variation in the imprint across the impact surface indicates the sharp differences in short impedances among the target materials. The wave in the quartz phenolic target and at the interface between the target and the C-7 epoxy will not be a smooth plane shock but will be "dimpless." As the wave propagates into the homogeneous C-7 epoxy, the faster, higher struck portions that enter the unshocked material first chould gradually be dispersed

The use of the term "appedance match method" is not completely justifiable here because the wave is neither planar nor steady state.

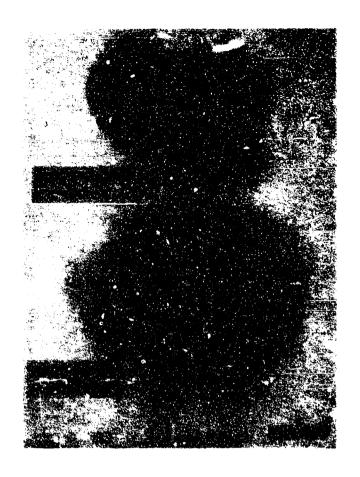


FIGURE 42 RECOVERED 3-D WEAVE QUARTZ
PHENOLIC TARGET AND ALUMINUM
GAS GUN PROJECTILE HEAD WHICH
IMPACTED IT. Impact velocity = 0.85
mm/µsec, Impact stress = 37 kbar.

by nonuniaxial flow. The wave in C-7 is thus expected to become progressively smoother as it propagates, which it apparently does (as indicated by the shorter rise time measured with the 0.080 inch gage). Further confirmation of the dimpled nature of the shock front is noted in the HE experiments discussed later.

The "effective" macroscopic average Hugoniot-type data measured by the manganin gages in the C-7 transducer material are displayed along with the quartz phenolic loading data in the stress-particle velocity plane of Fig. 41. It is evident from this figure that the gage stresses do not represent a reliable measure of the release states for the quartz phenolic as they should for steady state shocks. The disagreement is attributed to stress measurement uncertainties associated with the dimpled character of the wave impinging on the manganin wire stress gage:

- Local areas of the wire sense different peak stress levels arriving at different times. Hence, the actual level recorded at the oscilloscope corresponds to some average of the local stresses in the shock wave mesh traversing the wire at a given time.
- The resistance change in the manganin due to compressive stress (causing decrease in resistance) may be masked by local strains in the manganin caused by stretching of the wire (and an increase in resistance) as it is traversed in the lateral shock by the uneven wave front.

In Shots 13,421 and 13,422 there was some evidence of the possible formation of a precursor in the quartz phenolic with a wave speed of approximately 4.5 mm/µsec and a peak stress estimated from manganin gage data to be about 1 kbar. The stress record from 'he attenuation shot, Shot 13,422, is shown in Fig. 43. In traversing 9.22 mm of quartz phenolic, the shock stress attenuated from about 30 kbar to 4 kbar. The recorded rise time was about 1 µsec, i.e., longer than the rise time of the "Hugoniot" shots. The record shows a possible precursor at 0.6 kbar in C-7 and a reflection of the precursor off the main wave at 1.3 kbar.

Two HE experiments were performed on the quartz phenolic discs. The specimens were impacted by a high-velocity ( $\sim 3$  mm/ $\mu$ sec), 1/8-inch-thick 2024-T8 Al flyer to obtain both shock and rarefaction wave propagation information. Normal 4-ohm manganin C-7 gages (Ref.23) -(gage element 0.010 inch into C-7 from interface) were used to monitor the

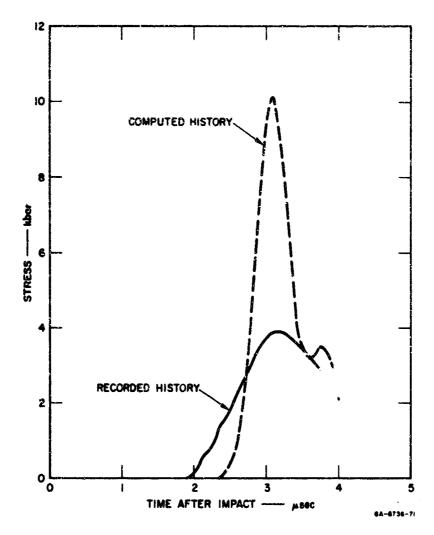


FIGURE 43 COMPARISON OF COMPUTED AND RECORDED STRESS HISTORIES FOR SHOT 13,422. 3-D quartz phenolic was impacted by a thin aluminum flyer; stress was recorded in C-7 behind target.

stress profile transmitted to C-7 by the quartz phenolic. The highspeed streak camera techniques described in Section III.2.b were used in the same experiments to record the specimen Hugoniot state and the shock ranarity. The manganin gage record, after data reduction, obtained for the 2.78-mm-thick specimen is shown in Fig. 44. Note that in the HE experiment the rise time is about 50 nsec and the peak stress fluctuates by more than 10 kbar, while in the low-stress gas gun experiment the rise time is about 500 nsec and the peak stress fluctuations are negligible. Peak stress variation (with a period of approximately 0.1 usec) which we attribute to the dispersive nature of the 3-D specimens are apparent and support the previous observation of nonsteady flow. The rise time of  $\sim 0.02~\mu sec$  was probably caused by the nonplanarity of the impact and therefore is not necessarily associated with dissipation mechanisms in the material. The rise times observed in the HE experiments are considerably shorter than those observed in the gas gun experiments because (a) the shock tilt problem is reduced at high projectile velocities; (b) at the higher stresses the shock transit times through both the quartz and the phenolic phases of the target are considerably reduced, resulting in less variation in absolute time of arrival along the gage; and (c) differences in the shock velocities of quartz and phenolic are less at the stress levels of the HE experiments than at those of the gun shots.

No evidence of a precursor is seen in the HE experiments. This observation is consistent with the results of the gas gun experiments, since at high stresses the observed shock velocity is considerably greater than the precursor velocity measured in the gas gun experiments. The quartz phenolic release data obtained from the C-7 manganin gages in the HE experiments have a large uncertainty (similar to that discussed for the gun shots), presumably because of the dimpled nature of the wave front transmitted by the 3-D weave quartz phenolic.

### 4. Theoretical Calculations

A limited number of calculations were made with the SRI PUFF 2 code using the usual equation of state for a solid (Eq.29) to model the woven quartz phenolic. One computation was made to simulate a Hugoniot shot and another to model the attenuation experiment. The purpose was to compare

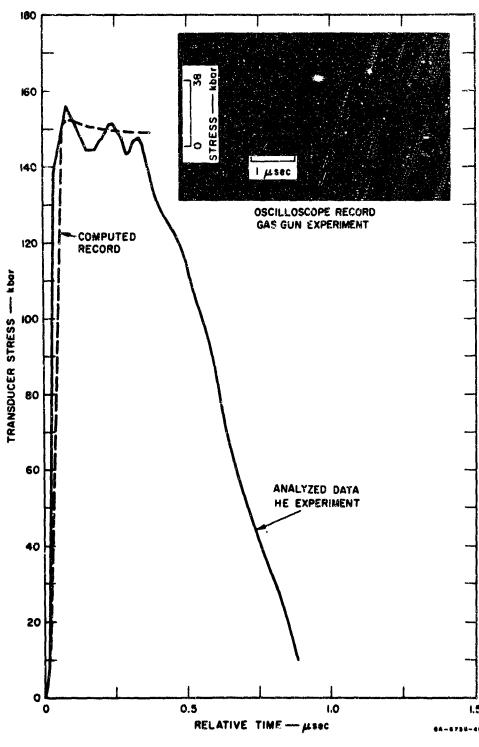


FIGURE 44 TYPICAL MANGANIN WIRE/C-7 TRANSDUCER RECORDS
OF STRESS-TIME PROFILES INDUCED IN C-7 BY HIGH-STRESS
(HE) AND LOW-STRESS (gas gun) SHOCK IN 3-D WEAVE
QUARTZ PHENOLIC

the peak stress attained, the rise time, and the arrival time. The comparisons were intended to show whether a satisfactory prediction could be made with the usual equation-of-state formulation. If not, the mputed results should indicate the types of modifications required.

The "effective" equation of state of the composite material was taken from the data of Fig. 40 to be  $U_s = 2.79 \pm 1.25 \ U_p$  (mm/µsec) and cast in the usual PUFF form:

$$p = 1.273 \times 10^{11} \mu + 2.167 \times 10^{11} \mu + 0.837 \times 10^{11} \mu^3$$

Yield strength was taken to be zero, and only normal \* values of artificial viscosity were employed in the computations.

The first calculation simulated the attenuation shot, No. 13,422, The 2024-T8 aluminum flyer was represented by the Bauschinger model with the parameters determined in Section IV. The computed stress history at the composite/C-7 interface is shown in Fig. 43 for comparison with the experimental record. The computed history shows significant attenuation from the computed impact stress of 37 kbar. The arrival time of the computed wave crest coincides with that of the experimental record, indicating that average wave velocities are correctly represented by the analytical equation of state. The most striking differences between recorded and computed histories are the overestimate of peak stress and the underestimate of rise time.

The portion of Shot 13,560 with the thinner specimen was also modeled by a computation. The stress history at the position of the manganin wire is shown in Fig. 44 for comparison with the measured history. The computed rise (governed by cell size and artificial viscosity) is about two times as large as the measured. The peak stress is comparable to the average but is not so oscillator.

A normal value of artificial viscosity for a computation is that which is sufficient to cause the total mechanical stress to follow a straight compression path (the Rayleigh lime) during a passage of a shock. For the present computations the coefficient of nonlinear viscosity  $C_0$  was 4.0 and the linear coefficient  $C_1$  was 0.2.

The following are probable causes for the discrepancies between the computed and experimental records:

#### 1. Computational effects:

- . Cell sizes used in the finite difference computations were 0.0001 to 0.001 inch, i.e., considerably smaller than the scale of inhomogeneity of the woven material.
- . Artificial viscosity was of normal magnitude, which usually spreads a shock front over four finite difference cells.

This combination of cell sizes and viscosity used in the calculations would normally lead to rise times longer than those observed experimentally for solids and did produce a longer rise time than that seen in the high-stress experiments.

## 2. Experimental effects:

- . The wavefront in quartz phenolic exhibited dispersion of the wave front associated with the variation of wave velocity between constituent materials. The oscillatory top of the stress wave at high stresses and long rise time at low stresses are indications of this dispersion phenomenon.
- . Frictional dissipation or material strength which increases with pressure is a possible dissipation mechanism.
- . Viscous damping is present in both the resin and the filaments (see Kolsky, (Ref.28) for example).
- . Concavity of the C-7 Hugoniot would lead to more rapid steepening of the shock front at high stresses than low.
- . A gage of finite dimensions has an averaging effect on a stress field with a spatial variation.

All three dissipation mechanisms—dispersive, frictional, and viscous—are present and may contribute significantly to the observed behavior.

The marked difference between the comparisons at low and high pressures leads one to suspect that some of the dissipation mechanisms are highly stress dependent. A general model for composite material should probably include all three dissipation effects.

# SECTION VII SUMMARY AND RECOMMENDATIONS

## 1. Summary

Four models for stress relaxation were selected and implemented in the SRI PUFF 2 wave propagation program. Their characteristics were compared with those of a general, or standard, stress relaxation model and discussed on the basis of several computer calculations. Dominant features of the computed results for all models were the attenuation with distance of the precursor and main wave and the extent to which these two waves were separated. Criteria were described for selecting a model and the model parameters.

A computational model representing the Bauschinger and related timeindependent effects was constructed on the basis of previous experimental and theoretical results and the present experimental data. The model was implemented in the SRI PUFF 2 computer program, and test runs were made to indicate the nature of the results obtainable.

An experimental technique and a ciated data analysis method were developed for obtaining complete loading and unloading compression paths in the stress-volume and stress-particle velocity planes. The analytical method was derived and applied to the present aluminum and titanium data. The method requires data from stress and/or particle velocity gages embedded at a series of locations in the target during an impact experiment. Valid results can be obtained for both stress-relaxing and rate-independent materials during Hugoniot and attenuation experiments.

Impact experiments were conducted on 6061-T6 and 2024-T8 aluminum alloys at stress levels of 15 to 600 kbar. Hugoniot data, release adiabats, and peak rarefaction velocities were obtained. Peak rarefaction velocities were considerably higher than those predicted by the standard elastic-plastic theoretical model; and the unloading adiabat was smooth, showing no break at an unloading point as predicted by the elastic-plastic model. Records at 50 kbar and 580 kbar were compared with computations made with the Bauschinger model. Conclusions drawn from the comparisons were that the model has the correct form, although some parameters could have been chosen more felicitously. No stress relaxation effects were

observed in the aluminum, probably because the samples were all thicker than those for which time-dependent precursor decay is important.

Impact experiments were made on samples of the common structural titanium alloy Ti-6Al-4V (mixed alpha and beta phases), type 50A titanium (pure alpha phase), and Ti-13Cr-11V-3Al (pure beta phase) in the range 15 to 750 kbar. Hugoniot data, Hugoniot elastic limits (HEL), and an alpha phase transformation data were obtained. The HEL in pure alpha and pure beta was double that predicted from static yield data, but the HEL in mixed alpha-beta was about as predicted. However, in the mixed phase material there was a large elastic-plastic transition ranging from 15 to 35 kbar. The phase transformation from alpha to probably ome, a phase at 50 kbar in the pure alpha metal showed only a small change in shock velocity and specific volume, and was overdriven at 62 kbar, suggesting that only about 10 percent of the material transformed at this stress. No transformations were observed in the other alloys for stresses up to 70 kbar.

Five impact experiments were made with woven quartz-phenolic to obtain preliminary Hugoniot data for stresses from 10 to 200 kbar. The records show an uneven wave front, oscillation in stress at the top of the wave, questionable release adiabat data, and a very thick shock front at the lower stresses. An attenuation shot with a 0.47 mm aluminum flyer yielded a stress reduction to 15 percent of the impact stress on the 9.46 mm quarts phenolic target, and a rise time of 1 µsec. A theoretical prediction for the attenuation shot based on a normal solid model gave much less attenuation and a steeper shock front. Evidently, highly effective dissipative mechanisms are operating in the woven material which have not yet been modeled theoretically.

## 2. Recommendations

The calculations with the Bauschinger model have led to significantly improved results for attenuation and unloading wave forms in two aluminum alloys; the model should be used whenever sufficient materials data are available for implementation.

The presently formulated Bauschinger model should undergo further development to make it more adaptable to other materials and for a wider range of stresses. Specifically,

- . more experimental data should be obtained on rarefaction velocities and unloading adiabats over a wide range of stresses in materials of structural and experimental interest, and
- the model formulation should be expanded to include a shear modulus and yield strength that are functions of specific volume and internal energy.

Testing should be undertaken in two areas which were only considered briefly in this report:

- . impacts on preheated specimens to map the equation of state and unloading relations at higher temperatures, and
- . reverse phase transition during unloading in materials undergoing phase transitions during compression. The multiple-gage analysis should be particularly useful in such a study.

The standard impact experiments a in woven materials do not yield strictly valid Hugoniot data because of the nonequilibrium states reached during the shock. The nature of the states achieved should be studied theoretically to indicate how to use the Hugoniot-like data for analytical predictions.

Experiments should be conducted to map the pseudo-equation-of-state surface for a woven material as a basis for theoretical predictions. The data could be obtained from impacts for a large range of stress levels, and with samples which have been

- , preheated to increase initial internal energy, and
- . manufactured with substantial porosity so that very large internal energies would be created during impact.

Stress levels used should include those required for vaporization.

An equation of state for woven materials should be formulated and implemented in a PUFF-type wave propagation code. The formulation should include frictional damping, viscous damping, dispersion caused by adjacency of materials with different wave velocities, and a delay mechanism such as stress relaxation.

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#### APPENDIX I

## TEST RUNS AND LISTING OF THE SRI PUFF 2 COMPUTER CODE

#### 1. Introduction

This appendix presents test routines and results that can be used to validate the use of the SRI PUFF 2 computer code on computers other than those at SRI. The code was developed to run on the CDC 6400 at SRI. It is a modification of SRI PUFF 1 to include stress relaxation and Bauschinger effects.

Any difficulties encountered in using the code on other machines will probably arise from differences in interpretation of some FORTRAN statements or possibly in running new types of problems. Usually these differences will trigger machine diagnostic statements or they will result in catastrophic errors. In the case of diagnostics, the error is located by the machine, and it can then be corrected. For catastrophic errors, the programmer needs some assistance in locating the source of the error. We have tried to provide this assistance in two ways: inclusion of data for test runs that will exercise all major portions of the code, and insertion of prints from each subroutine. The test runs include impact and radiation deposition problems and rezoning problems for solid and porous materials. The prints in each subroutine indicate the route taken through the subroutine (i.e., which branch of each IF statement is used) and list the main input and output quantities of the subroutine. These prints are triggered by presetting the constant MAXPR to the number of prints desired from the subroutines HYDRO, JSTRESS, HAFSTEP, EQST, POREQST, and SSCAL. Similar prints are obtained arom BAUSCHI, RELAX and BANDRLX by setting Serse Switch 3.

Six test problems are described in the following pages. The first problem has been run in three versions: a short test with triggering of prints from each subroutine, a short test with slightly altered data, and a full run with normal printout only. Some output from all three of these runs is given. For the other five test problems, only the full run was made. For all of the runs the major part of the output is given in the form of plots.

## 2. Input and Sample Output of Test Problems

## a. Test No. la

This problem, a test of the Bauschinger model implemented in the code, concerns the impact between a 6061-T6 aluminum flyer and a target of the same material. Behind the target is a manganin-glass stress transducer. The problem corresponds with a at 670 described in Section IV. This calculation is performed in three versions: a short calculation with printouts from many of the subroutines, a short calculation with printouts but using a slightly altered data deck, and a full run. The listing that follows is for the first of these computer runs. The listing includes the input data (which is essentially a reproduction of the data cards), the initial layout of the coordinate grid, test prints, the final EDIT, and the stress histories.

#### \*\*\*\* SRI PUFF 1 (6400 VERSION) \*\*\*\*

```
DATE #05/07/69
                                                         1 A SHORT TRIAL RUN ON DAUSCHINGED MODEL FOR SOCI-TO
 IDFNT 9
       COMPUTATIONS CORRESPOND TO THE 50 KBAR DATA OF ROBER WILLIAMS FOR 6361-T6 ALUMINUM. THE BAUSCHINGER MODEL IS USED FOR GOTH THE FLYER AND TARGET. THE JEDITS WERE PLACED AT THE APPROXIMATE THICKNESSES OF THE THREE THINNER SPECIFERS SO THAT THE COMPUTATION MODELS ALL FOUR EXPERIMENTS. THIS COMPUTATION WAS CONTINUED FOR ONLY FIVE CYCLES WITH PRINTS FROM EACH
        SURROUTINE.
O NCON =
 6861-TE AL FLYER
                                                                         RHG = 2.709E+00 NSRM = 5 NYAM = 1 NPOR=
 EGSTC = 0.070E-11 EGSTD = 1.000E-12 EGSTE = 1.220E-11 EGSTO = EGSTW = 2.500E-01 EGSTS = 0. EGSTW 9.000E-01 EGSTA EGST = 4.000E-00 C1 = 5.000E-02 -0.
                                                                                                                                                                                                                                2.949E+00
                                                                                                                                                                                                                                8.285E-12
  TENS(1) = -1.000E-11 TENS(2) = -1.000E-11 TENS(3) = -1.000E-11
                               6.000E+03 XP= 1.000E+00 MUUN # 6.676E+11 YADF = 3.210E+09 MU # 2.676E+11 YADD # 3.790E+10
                                                                                                                                                                                                                             1.000£+00
 IPA=
 Y0 =
                                                   25 CELLS IN 1.537E-01 CM DELXE +0.
  MIDNESH IN
 6091-YE AL TARGET
                                                                        RHO = 2.709E-00 MSRM = 5 NYLM = 1 MYOR = 0 NCON =
EGSTC = 6.670E-11 EQSTD = 1.006E-12 EGSTE = 1.220E+11 EQSTG = 2.500E-01 EQSTS = 0. EQSTN = 2.500E-01 EQSTS = 0. EQSTN = 0.00E-11 TENS(1)= -1.000E+11 TENS(2)= -1.000E-11 TENS(3)= -1.000E+11 TENS(3)= -1.000E+
                                                                                                                                                                                                                                2.0406+40
                                                                                                                                                                                                                                €.293E-12
                                                                                                                                                                                                                                1.9002+09
                                               210 CELLS IN 1.257E+00 CM DELX= -0.
                                                                           RHO = 2.520E+00 NSRM = 0 NYAM # 0 NPOR = 0 NCON = 0
 GLASS (MN GAGE)
 EGSTC = 8.780E+11 EGSTD # -8.780E+11 EGSTE = 3.000E+10 EGSTG = EGSTH = 2.500E+0: EGSTS # 8.750E+11 EGSTN 5.807E+00 EGSTA
                                                                                                                                                                                                                              2.40E+08
5.740E-12
 COSC = 4.000E-00 C1 = 5.000E-02 -0.

TENS(1) = -1.000E-11 TENS(2) = -1.000E-11 TENS(3) = -1.000E-11

NZONES= 1: 30 CELLS IN 3-150E-01 CM DELX = 6.000E-03
    EQST. J= 0. N= 0. CS9= 3.776E-11. EJ= 0.
                                                                                                                                                                   , DJ= 2.709E+00, PJB 0.
                                                           0: CSG= 3.776E-11, EJ= 0.
0: CSG= 3.484E-11, EJ= 0.
    EGST, JR 26. Na
EGST, JR237, Na
                                                                                                                                                                   . DJ= 2.709E-00. PJ= 0.
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TEST 1a: INPUT DATA

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TEST 1a: PORTION OF COORDINATE LAYOUT

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6E+11+ EJ= 0. . DJ= 2,709E+00, PJ= 0.
TO "1+962E-03 Y=2,14nE+09 TO 2,140E+09 COEF=-3,266E-03 MU=3,426E+1; DRO==1,42E-14 SDAV=-9,513E-84
                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                   EGGT: JE 29: No. 1: CSG# 3.776E-11: EJ# G. . DJ# 2.709E-00: PJ# 0.
No. 1 J# 29 [#2-2 SD# 0. TO -1.902E-03 YP?.140E-09 TO 2.140E-09 COEF##3.246E-03 HU#3.626E-11 DRO#-1.42E-14 SDAV#-9.510E-84
                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                 5.153E+06 5-017E+06
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MYDRO DTS 9.266E-09 RE 6.443E-10 SE 5.153E-06 CR 6.145E-05 DE 2.709E-00 TE-1.000E+11 LE 3.015E-04

MYDRO DTS 9.266E-09 RE 6.443E-10 SE 5.153E-06 CR 6.145E-05 DE 2.709E-00 TE-1.000E+11 LE 3.015E-04

DTS 9.266E-09 1.000E-12. U(J-1)E ... PHE 0. X#SE 1.597E-01 1.537E-01. CEFE 6.460E-05
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E0ST. Je 28. No. 1: CS6s 3.776E+11. EJs 0. . DJS 2.709E+00. PJR 0.
NJS 20 Es2. SDS-1.002E-03 TO -1.902E-03 Ys2.14-E-09 TO 2.140E+09 COEFs 0. NUB3.74
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TEST 1a: PORTION OF TEST PRINTS

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TEST 18: FIRST PAGE OF FIN AL TIME EDIT

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TEST 1a: STRESS HISTORIES

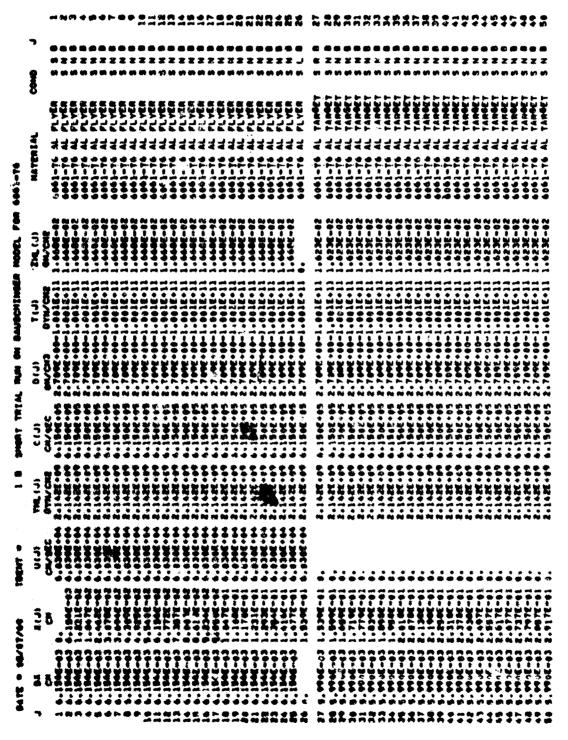
# b. Test No. 1b

This computation is identical to test la except that all the noninteger data are altered by 0.1 percent. The listing includes the input data initial layout of the coordinate grid, test prints, the final EDIT and the stress histories.

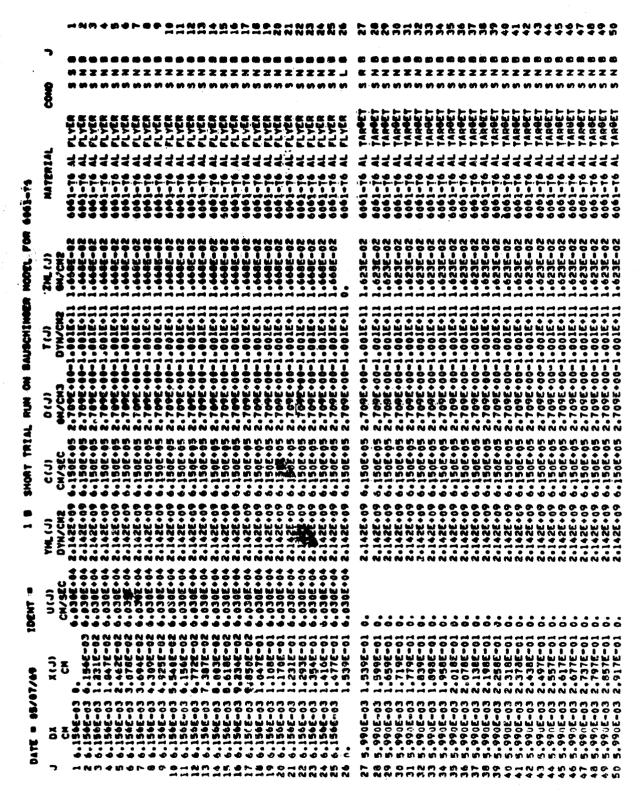
#### \*\*\* SRI PUFF 1 (6400 VERSION) \*\*\*\*

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DATE =05/07/69
 IDENT =
                        1 B SHORT TRIAL RUN ON BAUSCHINGER MODEL FOR 4061-14
   COMPUTATIONS CORRESPOND TO THE 50 KBAR DATA OF ROBER WILLIAMS FOR 6001-T6
   COMPUTATIONS CORRESPOND TO THE SO REAR DATA OF HOMER WILLIAMS FOR WISHING AND TARGET.
ALUMINUM. THE BAUSCHINGER MODEL IS USED FOR BOTH THE FLYER AND TARGET.
THE JEDITS WERE PLACED AT THE APPROXIMATE THICKNESSES OF THE THREE THINNER
SPECIMENS SO THAT THE COMPUTATION MODELS ALL FOUR EXPERIMENTS.
THIS COMPUTATION WAS CONTINUED FOR ONLY FIVE CYCLES WITH PRINTS FROM EACH
A MICUTE 7 NJEDITE 6 NREZONE 0 MSEPRATE 0 2 TEDITSE 3.009E-07 6.000E-07 9.000E-07 1.200E-06 1.609E-06 2.000E-06 2.400E-06 3 JEDITSE 50 80 128 172 239 251 6 NEDTME 100000 MEDITE
              MAXPR .
7 STOPS
                                                                                             3.000E-CA
8 NHTRLS=
                                                   1 UZERO .
                                                                 6.030E+04 IPLOT &
                        3 MATEL .
6061-TE AL FLYER
                                        2.709E+00 HSRM = 5 HYAM = 1 NPOR = 0 NCON = 0
                             RHO .
EQSTC = 6.677E+11 EQSTD =
                                        1.001E+12 EQSTE =
                                                                 1,221E+11 EQSTO =
                                                                                             2.94ZE+00
                                        0.
                                                                 9.886E-01 EQSTA ..
EGSTH #
             2.503E-01 EQSTS =
                                                    EQSTN =
                                                                                             8.285£-12
COSQ = 4.004E+00 C1 = 5.003E-02 C2 = 0.
TEMS(1) = -1.001E+11 TEMS(2) = -1.001E+11 TEMS(3) = -1.001E+11
XPO =
             6.006E+03 XP =
                                        1.001E+00 MUUN = 4.675E+11 YADF =
                                                                                             1.001E+00
             3.213E+09 MU =
                                        2.673E+11 YADD =
                                                                  3.794E+10
                      25 CELLS IN 1.539E-01 CM
NZONES= 1+
                                                                                            -0.
                                                                 -0.
                             RHO =
6061-T6 AL TARGET
                                       2.709E+00 NSRM = 5 NYAM = 1 NPOR = 0 NCON = 0
                                                                 1.221E+11 EQSTO = 2.042E+00
EGSTC = 6.677E+11 EGSTD =
                                      1.001E+12 EQSTE =
EOSTH #
             2.503E-01 EQSTS =
                                                    EOSTN =
                                                                  9.86E-01 EQSTA B
                                                                                             8.285E-12
                                        5.005E-02 C2 =
C050 =
             4.004E+00 C1 =
                                                                  0.
TENS(1) = -1.001E+11 TENS(2) = -1.001E+11 TENS(3) = -1.001E+11 XPO = 6.006E+03 XP = 1.001E+00 MUUN = 4.675E+11 YADF =
            6.006E+03 XP =
XPO .
                                                                                             1.0012.00
             3.213E+09 MU .
                     E+99 MU = 2.673E+11 YADD = 210 CELLS IN 1.250E+00 CM
Y0 =
                                                                  3,794E+18
NZONES# 1+
                                                                --0.
GLASS (MN GAGE) RHO # 2.520E+00 NSRM # 0 NYAM # 0 NPOR # 0 NCON # 0 EGSTC # 8.789E+11 EGSTC # -6.789E+11 EGSTE # 3.003E+10 EGST6 # 2.002E+00
             2.503E-01 EQSTS =
                                    8.789E+11 EQSTN -
EOSTH =
                                                                  S.ROIE+00 EOSTA L
                                                                                           5.740E-12
C050 =
             4.004E+00 C1 =
                                       5.005E-02 C2 -
                                                                  Ò.
TENS(1)= -1.001E+11 TENS(2)= -1.001E+11 TENS(3)= -1.001E+11 NZONES= 1. :30 CELLS IN 3.1936-01 CM DELX = 6.866E-03
NZONESH 1.
                                                                                            -0.
                      0. C59- 3.780E+11. EJ= 0.
 EQST. J. O. Na
                                                                   . DJ= 2.709E+00. PJ= 0.
 EQST. Ja 26. Na
EQST. Ja237. Ma
                     0. CSQ= 3.789E+11, EJ= 0.
0. CSQ= 3.488E+11. EJ= 0.
                                                                   . DJ= 2.709E+00, PJ= 6.
                                                                    , DJ= 2.520E+00, PJ= 0.
```

TEST 1b: INPUT DATA



TEST 16: PORTION OF COORDINATE LAYOUT



TEST 1b: PORTION OF COORDINATE LAYOUT

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|E-11; EJ= 0. . . DJ= 2.709E-80; PJ=4.799E-03 | MU=3.632E-11 DRO=-2.64E-14 SDAV=-1.905E-03 | TO -3.811E-03 | V=2.142E-09 TO 2.16ZE-09 COEF=-6.536E-03 | MU=3.632E+1 DRO=-2.64E-14 | SDAV=-1.905E-03 | MU=3.632E+1 | DRO=-2.64E-14 | SDAV=-1.905E-03 | MU=3.632E-1 | DRO=-2.64E-14 | DRO=-2.64E
                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                       $17-13
                                                                                                                                                                                                                                                                                                                                                                                                                                 ATH IN BAUSCHI 3 30

- OSTRESS NOWS MOURE QUE 4.447E-16 SDJ# 1.794E-06 F# 1.608E-08TJ#-1.601E-11 L# 0 0 50 0 57 0 0 0 70 0

- OSTRESS NOWS MOURE QUE 4.447E-16 SDJ# 1.794E-06 F# 1.608E-06 FJ# 3.270E-06

EDST: J# 27. Mm 1. CBG# 3.700E-11. EJ# 0.

- DJ# 2.192E-09 COEF# 1.746E-06 Y#2.142E-09 TO 2.142E-09 COEF# 1.746E-06 MU#3.794E-10 DROW 1.33E-05 SDAV#
                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                                           0 70
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JSTRESS NCRS MJRR MPJRO GJR 0. SDJR-3.811E-03 Fr 1.000E+001JR-1.001E+11 LR 0 0 50 0 57
KYDRO DTR 9.268E-09 RR-8.610E-03 SP-8.610E-03 CR 6.140E+05 DR 2.709E+00 TP-1.001E+11 UR 0.
CHR 0. DHR 2.709E-03 FMR-4.799E-03 MHR-8.610E-03 XHR 1.629E-91. LR 102
DTR 9.268E-00 1.000E-12. U(J-1)R 0. . XYSE 1.659E-01 1.599E-E1: CEPR 6.464E-05
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MAFSTEP, LB 28, NH 1 EMB 0. DAM 2.709E+00 PHS=4.799E-03 SDMS=3.811E-03 LB 0 0 0

CB 6.903E+05; CB 6.146E+05; CEFM 6.464E+05; COTM 4.004E+00; CITM 5.005E-02; RMM=8.610E-03

EGST, LB 28; NM 1: CSQM 3.780E+11; ELM 0. . DJM 2.709E+00; PJM-4.799E-03

MUM3.7
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E051. Jm 20. Mm 1. CS4m 3.796E+11. EJm 0.
m 1 Jm 20 Im2-2 50m 0. TO ~3.811E-0.
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26 HIJ:214
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TEST 1b: PORTION OF TEST PRINTS

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TEST 1b: FIRST PAGE OF FINAL TIME EDIT

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TEST 16: STRESS HISTORIES

## c. Test No. 1c

This computation represents the normal production run of Test 1. The results are a sample of those obtainable with the Bauschinger model. The listing included in the following pages contains the input data, one page of the initial layout, the final momentum EDIT, time EDIT, and two pages of the stress histories. In addition, there are computer-generated plots of deviatoric stress versus specific volume, of stress-distance curves, and stress histories.

#### --- SRI PUFF 1 (5400 VERSION) ---

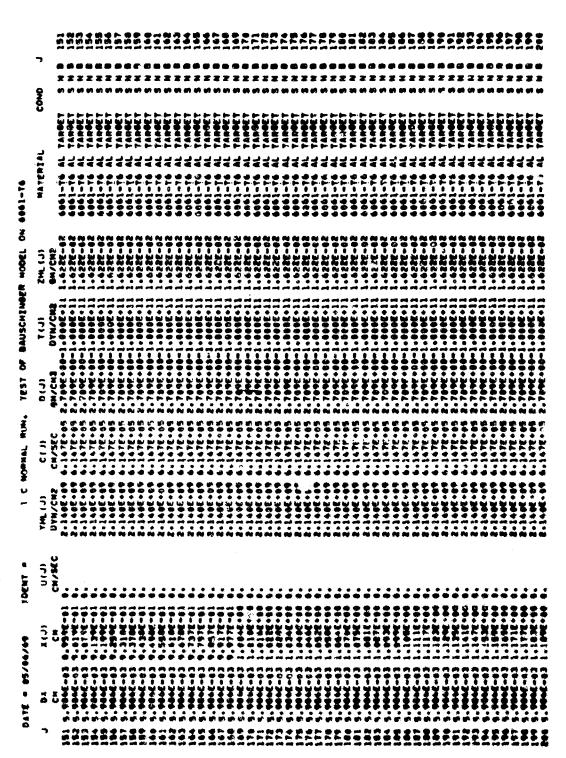
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DATE =05/06/69
 IDENT #
  TENT # 1 C NORMAL RUN. TEST OF BAUSCHINGER MODEL ON 6861-TG COMPUTATIONS CORRESPOND TO THE 50 KBAR DATA OF ROGER MILLIAMS FOR 6061-T&
   ALUMINUM. THE BAUSCHINGER MODEL IS USED FOR BOTH THE PLYER AND TARGET. THE JEDITS WERE PLACED AT THE APPROXIMATE THICKNESSES OF THE THREE THINNER
# CIEUTE 7 NJESIT# 6 NREZON# 0 NSEPRAT# C
2 TENITS= 3.000E-07 6.000E-07 9.000E-07 1.200E-06 2.000E-06 2.400E-06
3 JENITS= 50 80 128 172 239 951
6 NENTH# 100000 NEDIT# 10000 NPERN #
                                         1250 CKS .
                                                           3.500E+00 75 #
7 STOPS
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8 NHTRLS=
                                             1 UZERO . 6.030E+04 IPLOT .
                     3 MATFL #
6041-T6 AL FLYER
                            RHO = 2.709E+00;NSRM = 5 NYAM = 1 NPOR# .0 MCON & 0
EGSTC = 6.670E+11 EGSTD = 1.000E+12 EGSTE = 1.220E+11 EGSTG = 2.040E+00
                                   5.000E-02
EQSTH =
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                                                          9.893E-01 EQSTA
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C650 m
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           6.000E+03 XPm
3.210E+09 MU m
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EGSTH #
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TENS(1) = -1.000E+11 TENS(2) = -1.000E+11 TENS(3) = -1.000E+11 XPO = 6.000E+03 XP = 1.000E+00 MUUN = 4.670E+11 YADF = 7.670E+11 YADF = 3.790E+10
                                                                                   1-0005-00
                  210 CELLS IN 1.257E+00 CH DELX= +0.
NZONESE 1.
GLASS (MN GAGE)
                            RHO = 2.520E+00 NSRM = 0 NYAM = 0 NFOR = 0 NCON = 5
2.0002+00
                                                                                   5.740E-18
TENS(1) = -1.0008-11 TENS(2) = -1.000E-11 TENS(3) = -1.000E-11 NZONES = 1. 30 CELLS IN 3-150E-01 CM DELK = 6.009E-03
                                                                                  -6.
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TEST 1c: INPUT DATA

EST 1c: COORDINATE LAYOUT

TEST 1c: COORDINATE LAYOUT (continued)

TEST 1c: COORDINATE LAYOUT (continued)



TEST 1c: COORDINATE LAYOUT (continued)

TEST 1c: COORDINATE LAYOUT (continued)

TEST 1c: COORDINATE LAYOUT (continued)

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TEST OF BAUSCHINGER MODEL ON 6061-16	1(3)	DYN/CM2	.000E+11	.000E+11	.000E+11	.000E+11	.000E+11	.000E+11	.0006-11	.000E+11	. 000£+11	.000E+11	.0006-11	.000E+11	.000E+11	.0006+11	.000E+11	.000E+11	.000F+11	.000E+11	) )
TEST OF B	(5)0	6M/C43	2.520E+00-1	2.520E+08-1.000E+	2.520E+00-1.000E+1	2.520F.00-1.000E.	2.520E+00-1.000E+	2.520F+00-1.000E+	2.520E+06-1.000E+	2.520E+08-1.000E+	2.520E+00-1.000E+	2.520E+08-1.000E+1	2.520E+06-1.000E+1	2.520E+06-1.030E+1	2.520F+08-1.000E+	2.520E+00-1.000E+	2.520E+00-1.000E+1	7.520F+00-1.000E+1	2.520F+60+1.000F+1	2.520E+60-1.000E+1	
I C NORMAL RUM.	(2)	CM/SEC 6	5.903£+05 2.	5.903£+05 2.	5.903E+05 2	•	5.9038+05 2								5.903E+05 2.						
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TEST 1c: COORDINATE LAYOUT (concluded)

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TEST 1c: FINAL MOMENTUM EDIT AND PORTION OF TIME EDIT

TEST 1c: FINAL TIME EDIT (continued)

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TEST IC: FIRST PAGE OF STRESS HISTORIES

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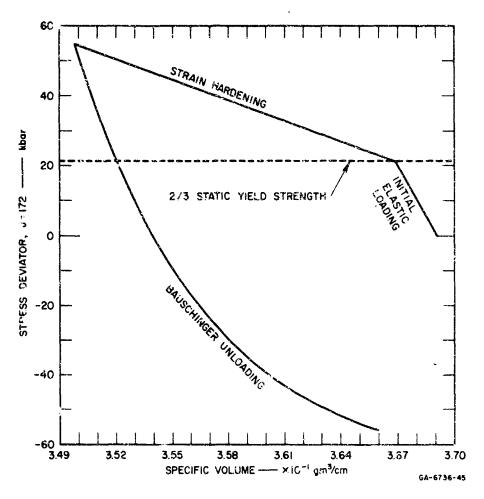


FIGURE 45 TEST 1c: SAMPLE PLOT OF STRESS DEVIATOR (Curves plotted by computer from calculated results)

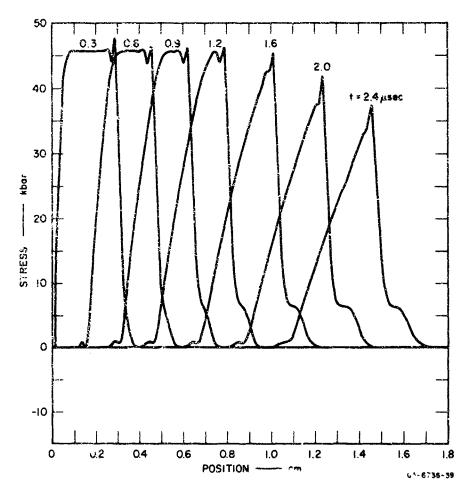


FIGURE 46 TEST 1c: SUMMARY PLOT OF ALL TIME EDITS

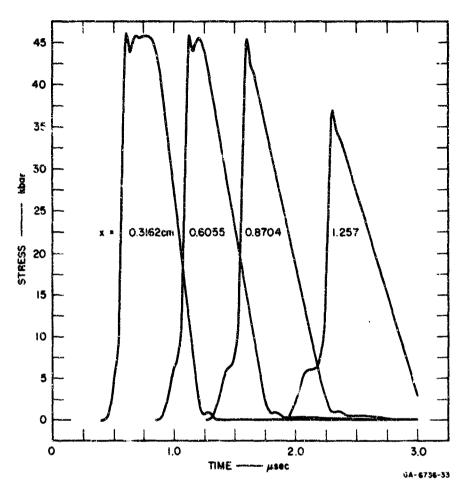


FIGURE 47 TEST 1c: SUMMARY OF STRESS HISTORIES AT FOUR JEDIT LOCATIONS

# d. Test No. 2

A simple impact between a solid aluminum flyer and an aluminum target is given as Test 2. Neither stress relaxation nor Bauschinger effect is included, so the problem could be run with any one-dimensional wave propagation code containing an elasto-plastic equation of state. The listing in the next pages includes the input data, a page of the initial layout, and samples of the final EDIT and stress histories. The EDITs, stress histories, and a deviatoric stress-volume path are shown in the figures.

# \*\*\* SRI + . F & (6400 VERSION) \*\*\*\*

```
DATE =04/22/69
DATE = 04/22/69

IDENT = 2 MODEL 0+ AL/AL IMPACT+ NO RELAXATION OR BAUSCHINGER

FLYER AND TARGET ARE BOTH OF 2024-TB ALUMINUM

1 NTEUT= 5 NJEDIT= 6 NREZON= 0 NSEPRAT=

2 TEDITS= 5.000E-07 1.000E-06 1.500E-06 2.000E-06 2.500E-06

3 JEDITS= 38 75 112 187 224 261

6 NEDTM= 100000 NEUTT= 100000 NPERN = 1

5 TOPS JCYCS = 1250 CKS = 4.000E+06 TPLOT = 2.86
                      JCYCS ...
8 NHTRLS=
                                                  1 UZERO =
                                                                   2.000E+04 IPLOT =
1 NPOR
                                                                                           9 NCON - 4
                                                                   1.220E+11 E9576 .
                                                                                              2.040E+00
                                                                                              7.5256-12
                                RNO = 2-785E+00 NSRM = 0 NYAM . 1 NPOR
AL TARGET
                                                                                           0 NCGN = 0
2.940E+90
                                                                                              7.525E-12
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TEST 2: INPUT DATA

EST 2: PORTION OF COORDINATE LAYOUT

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254 2 ENVPOS 1146E+05 3 X (LEMBO3)	2.802586 P 074/C	-66 6-914E -07 1.698E -07 3.403E -07 6-353E	+08 1.9235+01 +08 3.1356+01 +08 4.8965+01 +08 7.3216+01	1.4386 1.8896 2.8826 3.8966			~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~ ~	64 4.1566 64 1.1566 64 1.1566 10 1.2566 10 1.3566 11 1.3566
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TEST 2: FINAL MOMENTUM EDIT AND PORTION OF TIME EDIT

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7			•			۰	•	0.000		፥	:	2.2	2.
3	.00	19.707		•		0.000	-	•	9	:	167	٠	~
: 3	1	15.797	••••		0000	0.00.0	3	6.000		÷	167	•	2
<b>`</b>	•	,    -	•		•								

TEST 2: FIRST PAGE OF STRESS HISTORIES

TEST 2: STRESS HISTORIES (continued)

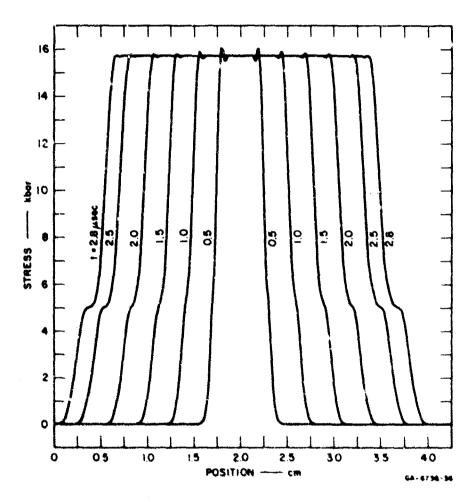


FIGURE 48 TEST 2: SUMMARY PLOT OF TIME EDITS

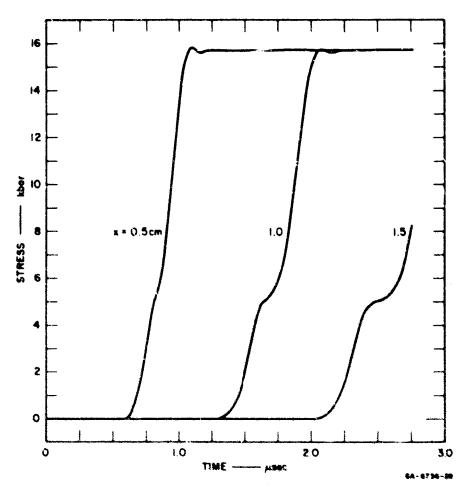


FIGURE 49 TEST 2: SUMMARY OF STRESS HISTORIES AT THREE JEDIT LOCATIONS

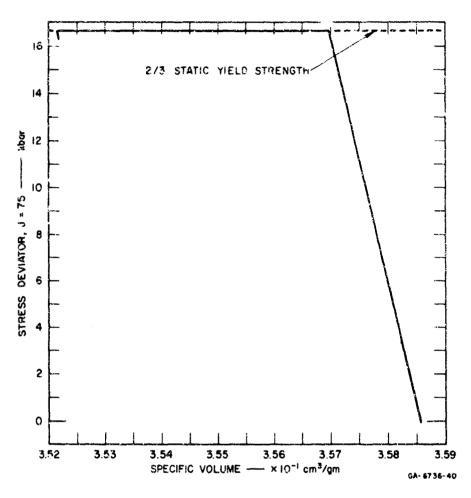


FIGURE 50 TEST 2: SAMPLE PLOT OF STRESS DEVIATOR

# e. Test No. 3

The stress relaxation sechanism of Model 1 (simple anelastic model) is tested in Test 3. The configuration is a symmetric impact between two plates of 2024-T8 aluminum. The plates have different relaxation parameters but are otherwise identical. The listing included on the following pages contains the input data, and portions of the initial layout, and final EDIT and stress histories. The EDITs, stress histories, and a deviatoric stress-volume path are shown in the figures.

#### \*\*\* SHI PUFF 1 (6400 YERSION) \*\*\*\*

```
DATE #04/25/69
4.000E+00 T5 =
2.000E+04 IPLOT =
7 STOPS
               JCYCS #
                              1250 CKS =
                                                              2.600E-64
8 NHTHLS=
                    RHOS = 2.7651-00 NSRM = 1 HYAM = 1 HPOR = 0 NCON =
AL FLYER
        E051C =
                                                              2.0402.00
6050 = 4.0002.60 Cl = 1.5001-01 CR = 0.
TENS(1)= -1.000E+11 TENS(2)= -1.000E+11 TENS(3)= -1.000E+11
TRLX a 1.500E-07 -0.

YO = 2.500E-09 MU = 2.670E-11 YADD M

NZONES= 1. 148 CELLS IN 2.000E-00 CM
                                           eû.
AL TARGE!
                    RHOS = 2.7856-80 NSRH = 1 NYAM # 1 NPOR = 0 NCON = 0
2.040E-00
                                                              7.525E-12
TRLX = 3.000E-07
YO = 2.500E-09
        2-5008-09 MU # 2-870E+11 YADD #
148 CELLS 1H 2-0008-00 CH
MICHESE I.
                                           -0.
```

TEST 3: INPUT DATA

TEST 3: PORTION OF COORDINATE LAYOUT

-
-
0
ũ
3
-
3

	2.4073E-04 01PULS	1.4060L-0 EXYM	w	111 2.69 HVPOS	6513E+01 EMVPL	299 Emvpr	NA PE		101 Emvum	1.5011E+10 POTPOS	2 - 3404E - 05 F012E 6	
7.1444E-96 3411 3.6165E-92	3.2379E-06 X (JBNU1) 2.0279E-0	6. X (JBH02	1.1.1 2.1.1.0	N 4	· · ·	. 176	1 • 1296E • 05 RMAX 1 • 5812E • 10	• × •	X (JRKAX) 3908E+U0	•	•	
TIME EDIT NO.	1 MO. 6 AT	N 8 237, TIME		•90726E-06	SECS. JSTAN	IAN # 248.	CALC TIME	15 14	149.922	secs		
		3	a	•	Ĭ			7	CONO		T. NM. NU SU	0
	1		0.147.042	CHOUNT	7.90.J	6K/CH3	CM/SEC CE				I DYN/CM2 OR	_
, , ,			,		386 40	2.7850 B	39425 + 05	5	4	S. VER	1.6672.09	1.8925+0
	4000	0.00 PART 1.00 P		1.5106007	47.4	2.785± 6		G	¥	>	1.6676-19	6.592E 00
4 L L L L L	9266		2.45454.5	7.79	•	2.783. 6		_		FLYER	1.6676+39	1.428E+07
344	1000		• • •	7-4015-07		2.7852	3918E+05	S	¥ E	FLYER	•	2.780E+0
110197	71701		V-142E+07	3776	-	2.7853	.3921C+05	S	9 AL	6 LYEA	2.667£ 0.09	S - 1 - 10 E + C
143708	15961			2.439E+08	1.2276+0	2.7856	.39278	v	¥ ¥	FLYER	Ö.	8.9645.40
137218	19759			7		2.7860	.3936E+05	V)	8 4 6	FLYER	3	1.5005.
1507051	14612		90	×	B. 7	2.7866 6	.39506+05	v	æ ₩	FLYER	3	2.4175.00
10022	1946		30	1.027E+09	2.036E+05	2.7875 5	.3970E+05	y)	¥	FLYER	2 - 66 7E + 09	30000 P
111727	17130		60.	1.508E+09	.284E+0	2,7887 6	2	Ś	# #	F 45.00	2.667E+UV	9.375K+0
.171219	16769		60+	1076	æ	2.7901 6	•	(r)	۔ ج	FLYER	•	0+1890+
.Z.4704	18400		*0*	2.199€+09	_	2.7918 6	3	S	¥ .	FLYE	300 / W 1 / V	
-21018	17986	_		.535E	N	2.7436 6	. 4 106E + 05	'n	€	FLYEN	AC+11000	0+3 <b>4</b> 22-
323344	1 17543		66	4.250E: 09	m	4.7954 6	6.41456+05	un:	ď. Œ	FLYER	<b>?</b> :	1043/6401
.2*51c	7 17252			4.874E+09	n	2.7969 6	.4174E+05	Ś	<b>₹</b>	FLYEN	AD+2/00**	
.23456	76591			5.3526.09	•	5.7980 6	* 4 205E + 05	v	4 V	F 1 2 5 8	アフマルトののの	1 - 7 9 9 E + 0
.272909	16766			5.717E+09	5.3896.0	2.7390 0	•	S)	¥ .	FLVER	1000 1 000 1	0.3004.
.205452	16596	•	50	6.038E0.0	S	6.7998 6		S (2)	₹ :	FLYEK	1.447F4.0	3 - 34 - 6
.24884	3 16415		9	. 34 2E	6.7482.0	0 /000-7	•	n			300 12 44 1	2.104540
.312328	16233		e :	- 6+ 0E	-	9 9708.2	7	<i>n</i> u	4 4	M	1.6676+09	2.1556+09
30167F.	00001		2 6	40F4 0	8.2002.00	4 66 66 4		n <i>v</i>	\ \ \ \	Me A	1.5672.	2.2026+0
25.4	CORCI	1 10 1 10 1 1		7 5235.09	• 0	7 40 20 7		, v	¥	FLYER	1.667E+.19	2.244E+09
019266.	9796	1	3	7 H 325 + 00	٠,-	2.865	43648+05	S	4	FLYER	1.6678.09	2.281E.0
746/6	15.247		9	•	٠-	Z - 800 J		S		FLYER	1.667E+09	2.315E+0
3 % > # 5	47			4346	۰ -	2.6071 0	*	ທ	A AL	FLYEM	1.6678+09	2.3456.0
	14011	•	30		1.+u1E+07	4 1808 2	*	S	A AL	FLYER	1.6675+04	2.371E.09
41965	21/41	•	60	9.034E+09	-	2.8091	*4*51E+05	٠.	A AL	FLYEH	1.667E+39	2.392E.0
44305	3 14523		6	9.331E+09	-	2.8101	6.4473E+05	(A)	₹ •	FLYER	1 - 66 7E+JY	2.410Ee0
	14330	`	3	.6256	~	2.8110	.44956+05	s	- W	FLY5K	90+31-96+4 00+31-96+4	2.426E.009
4286e+	14140	ر ب	50+	1 SE		2.8160	04517E+05	v: c	¥ .	ال ا	001001001	2-4476
.4/321	13452	•	٠	₹.	2.004E+07	9 06 18 9	•	n 1		בי גנט	000000000000000000000000000000000000000	2.45.35.40
85024	13766	30 : 30 :	. U30E+09	7 -	2.132E+07		0043C0040			# K F F	90 4 3 K 4 5 4 K	2.6556+0
	, .	0 I	3016.04	1.05.	70.300.0			) U	i 4	 	1-667E+09	2.455E+09
51132	~ ,	D 1	2	1 + 3201 •	• -	4 14 18 18		n u		2	1.06677-39	2.452E+0
.9696		) 	,	•1685 • 1	0.3576.	4 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6	•				1-6678+39	2.447E+0
. 40044	٠.	) )	5	133641	7000			, ,	<b>*</b>	7 A F R	1-6670+09	2.4395.0
	•	<b>)</b>			•				` <b>4</b>	1 de 1	1.66 2.09	2.429E+0
		, ;	3000	3.5011.10	٠.	( MIC 20 )	7	•	¥ .	FLYER		2.416E+09
	7 .		• •	2446	٠	2	-	, v	\ \ \	FLYER	. •	2.402E+0
	~	~ >	٠.		70.00	4 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	,				000	2.385E+0
		•				֡						

TEST 3: FINAL MOMENTUM EDIT AND PORTION OF TIME EDIT

			1.673E+09
999845999999 98999999999			1.667E+09 1.667E+09 1.667E+09
	4 - 4 - 4 - 4 - 4 - 4 - 4 - 4 - 4 - 4 -	東京の音音を表現である。  東京 はままま はままま はまま はまり はん 人人	<b>4 4 4</b>
6 6 4 4 7 7 2 6 6 6 6 4 4 7 7 2 6 6 6 6 6 4 4 7 7 2 6 6 6 6 6 4 4 7 7 2 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6		6 - 5 - 5 - 5 - 5 - 5 - 5 - 5 - 5 - 5 -	6.5046F-65 99 6.5046F-65 100 6.5046F-65 100
	20 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2		.#616.47 2.835 .#666.67 2.835 .8746.87 2.635
70E*10 1.305E*1 20E*10 1.323E*1 20E*10 1.354E*1 20E*10 1.354E*1 20E*10 1.398E*1 20E*10 1.405E*1 20E*10 1.45E*1 20E*10 1.45E*1 20E*10 1.45E*1 20E*10 1.45E*1			*10 1.581E+1
016 1.3096*ic 1.9096*ic 1.9096*ic 1.9096*ic 1.9096*ic 1.9096*ic 1.9096*ic 1.9096*ic 1.9096*ic 1.9099*ic 1.909*ic 1.909*i	915 1.*6.26.10 1.*  944 1.*  9	1 1 5 5 6 6 6 6 1 6 1 6 1 6 1 6 1 6 1 6	995 1.581E-10 1. 994 1.581E-10 1.
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rest 3: FIRST PAGE OF STRESS HISTORIES

	SCTPUT.	TIME IN MIC	CHOSECS. S	IF KBAPS.	DIME IN	DIMM IN NAMOSECS. DELIIM IN		SECS					
*	7 1 1 1 1	\$12/2013	St. 087 'S	\$34/\$0.13	\$ ( 30)	\$ (75)	\$(112)	\$ (187)	\$ (224)	\$ (261)	213	DTREE	DELTI
ī	4	14.164		979		9000	6000.	6.0000	•	0.00	173	1.600	2
: 2	795		3010	0.00	8000	3006.0	3000	900000	8000.0	6000.0	?	1.931	1.63
13	<i>*</i>	16.150	•				3000	9000		0,0600	**	2.317	2
S	- Pro-	16.154			:		8000.	9.1000	•	600° .0	-	2.701	2.
3	316	16.150			:		. 000			F003.0	*	J. 3.	.2
1	-81	10.245				90000	8900.	0.000		3000 Q	•	4.96.4	<u>ٿ</u>
-	114.	16.1.9		00.0			3000	% <b>0 • 0</b>	•	900000	146	4.005	2
3	200	16:131	***	0000	0.00.0			****	•	8000°C		<b>5</b> °.√ <b>\$</b>	Ý
\$	. 532	16-123					9000.	900.	•	000.0	£ 3	6.919	Ž.
\$	3	16-113		0.00	:	£.00.3	9000	300.		00000°0	3	9.363	ě.
***	980	16.102	••••	•	•	6.00£		• 000	90.00	Q		44.6	9
3	3	\$	•	900	0000	0.6000	*000.	9009	•	100°0		13.957	.3
2	. 976	16.075	0000	ě	9000	90000	90000	8000		0.00 to		14.056	
I		1	٠	0.00	3	0.000		. 0630	•	80.0.0		14.057	.3
1	1	16.053	3	0.00	6	0.00	.0028	.0628	•	9969.0		14.057	~
Z	914.	13.642		0.00	•		.0078	. 0.076		0.000		7 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	7
2.0	77.	16.032	ĕ	•	:		. 62	.0210	•	8000. 6		4.657	R5.
3	*	10.023		•	3.000	0000.0	8750.	. 1550	0.00.0	0 - 44 - 7		14.657	ď.
\$	Ş	16.084			:	200.0	1298	201.	•	0.000		14.057	¥
	. 673	10.04		•	600		.2000	.2978	74	8.000		34.657	7
-	į	15.9				200.3	20105	3604	•	0.000	-	14.057	1
7.5			•		9000	00000		1.1.2		80.00	3	14.057	¥.
2	717	15.903			000	4.000	1.6528	39.601		0.00.0	841	14.057	÷
	121	15.576			:		2.910%	3.1470			641	14.057	Ę.
K	3	15.960			3	9.00.6	4.2u2B	4.6190		0,000	3	16.057	F.
		18.42				2000.0	5.585A	6.2310	0.000	0.00.0	=	14.057	7
7	.172	18.957	9.10	•	00000		4+06·9	7.462	•	0.0000	3	14.857	÷
E	. 707	19-61		****	7	2000.0	8.097A	4.377A	•	9.00	**	14.057	¥.
Z	3.	19.00	0.00	•	90000		9.1+0A	10.6964	<b>2</b> • • • •	6.0.0	2	4.057	<u>ب</u>
3	\$19.	19:41		0 0 ° D		20.0	10.0164	11.7444	•		3	10000	Ť
=	į	15.135	• • • •	•	••••	200,0	19.7147	12.526A	•		•	100-1	*
3		15.020	::	****	30000	00000	11.254A	13.068A	•	000,	9	14.057	•
3	.63	18.924	9000	****	:		11.6.11	13.426A	0000				
3		16.018	=	•	9000	7	11.9964	130007	7				
3	Ş	210-61		•	2		16.64.6.	13.734					
8 4							1000	13.04.61				750	
: 1							12.8604	14.6134		9000			•
3	7	19.645		9	2		13.026A	30.068A	7.8668	0.00.0	:	•	•
z	į	15.077			2		13.1024	14.104A	7	0.000	=======================================	•	*
- T	. 970	15-072	•	0.00	:	7	13.530A	14.149A	₹	0.9086	=	•	*
3	į	18.84	•		3	****	13.469A	14.186A	•	9. 6.50	2	4.000	Ž.
\$		15.547	٠	••••	•	8	13.601A	14.225	ï	0.000	•	450-41	
I	1.000	12:071	-00-5		9.000	ë	13.06.7A	14.266	7	3	14	2.133	•
8	1.063	15.071	•	•••	0.000	3	13.607	14.2740	₹	3	=	7.934	
#		12 0 0 M	•			6	13.712A	14.200A	•	2	7	3.072	
5		12-67	•	99.0	Ö 1	9.000	13.7484	100% • V	•	•			
2	7007	12-071			•	3	HOLL OF A	14.C.	₹.	2	B 1	746.	
<b>S</b> ;	1.019			000			4619.61	400F-47					
¥	ë.	2/2·61					140000	1996 • 1			•		}

TEST 3: STRESS HISTORIES (continued)

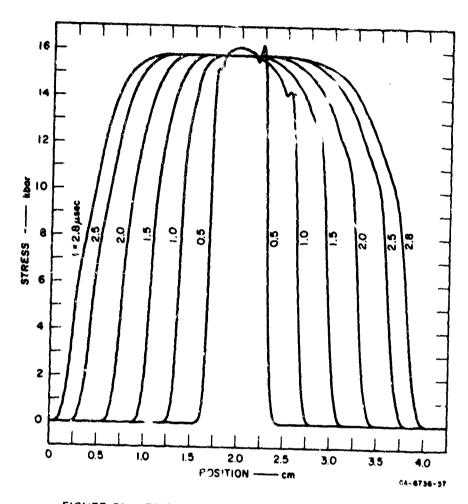


FIGURE 51 TEST 3: SUMMARY PLOT OF TIME EDITS

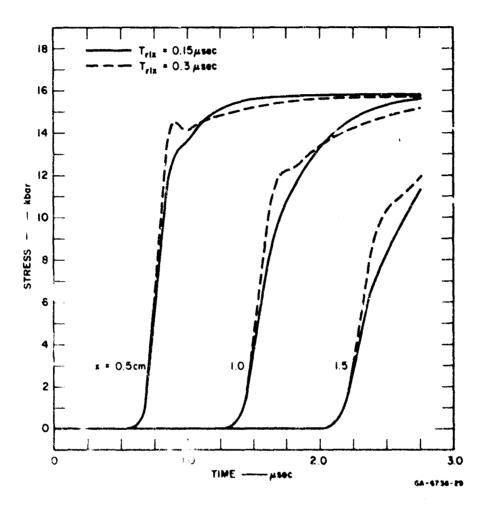


FIGURE 52 TEST 3: SUMMARY OF STRESS HISTORIES AT SIX JEDIT LOCATIONS

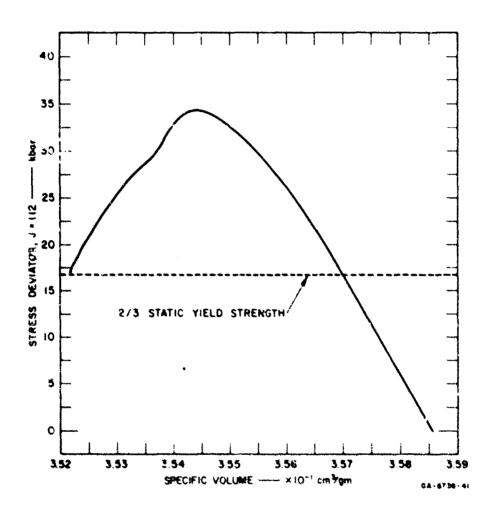


FIGURE 53 TEST 3: SAMPLE PLOT OF STRESS DEVIATOR

### f. Test No. 4

The stress relaxation mechanism of Model 2 (Band's model) is tested in Test 4. The configuration is a symmetric impact between two plates of 2024-T8 aluminum. The plates have different relaxation parameters but are otherwise identical. The listing included on the following pages contains the input data, and portions of the initial layout, and final EDIT and stress histories. The EDITs, stress histories, and a deviatoric stress-volume path are shown in the figures.

### \*\*\*\* SHI PUFF 1 (6400 VERSION) \*\*\*\*

```
DATE #05/13/69
                4 MODEL 2. BAND RELAXATION MODEL. IRON/IRON IMPACT
IDEAT =
1 MAXPHN = 4.000E+00 TS =
7 STAPS
                 JCYCS =
                               1210 CKS =
                                                               8.000E-07
2 MATEL =
                                  1 UZEHO =
                                             2.500E+04 IPLUT #
A NMTHLS
                     HHOS = 7.850E+00 NSRM = 2 NYAM = 1 NPOR = 0 NCON = 0
                                 EQSTE = A.400E+10 EUSTG #
                                                              1.500E.00
                                             1.980E+10 VM =
                                                               3.220E+05
                                             5.000E+00 NTO =
                                                              -0.
IRON TARGET
                     RHOS = 7.850E+90 NSRM = 2 NYAM = 1 NPOR = 0 NCON = 0
                               EGSTE = R.400E+10 EGSTG =
EQSTC = 1,741E+12 EUSTD = 0.
                                                              1.600E+00
EGSTH = 2.500E=G1 EGST3 = COSG = 4.000E+00 C1 =
                                                               7.214E-12
                          0.
                                             1.650E+00
                           1.500E-01 C2 #
                                             0.
 TENS(1) = -1.000F+11 TENS(2) = -1.000E+11 TENS(3) = -1.000F+11
       1.000E-06 T2 =
0. EP5 =
                           4.000t-06 HEE =

9.000t-01 NMO =

8.14Gt+11 YADD =
                                                               3.220F+05
                                             1.980E+10 VM #
         0. EPS
5.000E+09 MU =
                                             S.000E+00 NTO =
                                                               2.000E+01
GEE =
                                             0.
YO =
             148 CELLS IN 6.0001-01 CM
NZONES= 1+
```

TEST 4: INPUT DATA

	7		60	an	<b>6</b> 0	<b>3</b>		<b>æ</b>	<b>(1)</b>	<b>3</b> 2	Œ	•	œ	æ	•	<b>æ</b> :	<b>æ</b>	<b>aa</b> , ,	<b>a</b> n (	<b>30</b> (	B (	9 0	، ۵	<b>a</b> 4	70 G	0 9	D Ø	0.0	0 45	<b>a</b>	<b>a</b>	<b>GD</b>	<b>a</b>	<b>20</b> 6	P 00	<b>3</b>		•	D 4		<b>a</b>	æ	60	æ	<b>a</b>	<b>10</b> , 0	D 4	B 62	,
			s	z	2	z	7	z	¥	2	2	z	z	z	z	z	z	z	z	z	<b>z</b> :	<b>.</b>	<b>z</b> :	Z	Z 2	2 3	2 2	2 3	. 2	. Z	z	z	Z	z :	2	z	2	Z.	2 2	. 2	22	z	z	Z	Z	<b>z</b> :		2 7.	:
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PACT	MATERIAL		IRON FLYER			IRON FLYER		_		**	IRON FL. YEA		_														TRON FLYEN		TEON FLATER		IRON FLYER			IRON FLYER	TEON FLYES		_		MAN FLARM			IRON FLYER				THON FLYER		ESTATE NOOT	
MODEL 2. BAND RELAXATION MODEL. IRON/IRO: IMPACT	2ML (J)	GK/CH2	3.182£-02	3.182E-02	3,1825-02	3.182E-02	3.182E-02	3,182E-02	3.1825-02	3.1825-02	3.1825-02	3.182E-02	3.1826-02	3.1826-02	3.182E-02	3.1326-02	3.182E-02	3.1826-02	3.182E-02	3.182E-02	3.182E-02	3.1865-02	301328746	3.182E-02	3,1825-02	3.15CE=02	3.1925-02	30 4 2 4 6 5 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6	3.1825.02	3.182E-62	3.182E-02	3.182E-02	3.162E-02	3,182E-02	3.1825-02	3.162E-02	3.1825-02	3.1025-02	3.182E-02	3.183F-02	3.182E-02	3.182E-02	3.182E-02	3,1825-02	3.182E-02	3,162E-02	3.10ct0c	3.1825-02	->
UN MODEL.	3,	DYN/CM2	.850E+00-1.000E+11	. ASOE +00-1.000E+11	-1.000E+11	.850E+00-1-000E+11	.850E+00-1.000E+11	.850E .00-1.000E -11	-1.000E+11	. #50E + 00-1.000E+11	.850F.06-1.000E.11	-1.00 -11	.850E+86-1.06vE+11	.850E+0C-1.000E+11	-1.600E+11	.850E+00-1.000E+11	.850E+00-1-000E+11	-1.000E+13	-1.000E+11	. 850[ +30-1 .000E +11	.850E+00-1.000E+11	- #50E + 00 = 1 • 900E + 1 E	Section 1 - 00 - 10 - 00 -	. A50E+00-1-800E+11	.050E+00-1.000E+11	. MSGE + GO - I + UGGE - I I	.850E+00-1.000E+11	11-2000-1-00-3050-	. #306 + 10 - 1 + 0 0 0 E + 1 1 . #50 6 + 0 0 - 1 + 0 0 0 E + 1 1	. 6-00-1-000E-11	.850E+00-1.000E+11	.850E -00-1.600E+11	.850E+00-1.000E+11	. #50E+00-1.000E+11	. 550E + 00-1 + 000E + 1 L	.850E+00-1-009E+11	.850E+00-1.000E+11	-1.000E+11	**************************************		A50E+00-1,000E+11	. 850E+00-1-000E+11	. ASOE + 00-1 . 000E + 11	.350E .00-1.005E .11	.859E+00-1-000E+11	.850E - 00-1 - 000E - 11	20000	/.850E+00-1.850E+1.	1
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0 1	7HL(J)	CMUNICHO	3,333E+09	3-333E+09	3.3336+09		3.3338+09	3.373E+09	3.3335.09	3.3336.09	3.3336+09	3.3335+09	3.3336 + 09	3.3338+09	3.3336+09	3.333E+09	3.3336+09	3.3336.09	3.3336+09	3-333E+09	3.3335.09	3-3335-09	3-3332-04	3.3335.09	3.3335.09	3.3332.09	3.3336.09	3.5336.09	3 - 3 3 3 E - 0 4	3.3336.09	3.3335.09	3.335+09	3.3336.09	3.3335+09	3.3335.00	3.3336+09	3-335-09	3.3335.09	3-3335-09	353556	3.3335.69	3+333E+09	3.3336.09	3.3335+09	3-3336-09	3-333F +09	3.3335.09	3.3335.00	
IDENT .	25	J95/MJ	2.500E+04			N	N	~	N	N	N	~	N	N	N	N	N	N	~	N	ċ	~	N	N	N (	<b>V</b>		2.500E-U4	Z.500E.04	2.500E+04	2.500E+U4	2.500E+U4	2.500E+04	2.500E+04	2.500E+U4	2.500E+04	2.5006+04	2.50CE+04	2.500E+04	2 500E + 114	2.500E+U4	2,500E+04	2.500E + 04	2.500£+64	2.5006.04	2.500E+04	Z.500t.04	Z.500E.04	
.5/13/69	í) x		é	•					N	_	~		_			-		-		•	_			₽ .	<b>6</b>		-	٠.	1.0546-07	~ ~	-	-	~~	-	1.3386-01	• ~	~	<b>-</b>	→ -	1.3815-01	-	•	-	1.7845-01	-	1.865E-01	1.9055-01	1.9465-01	1004.1
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TEST 4: PORTION OF COORDINATE LAYOUT

K(JSMAX) 4.5575E-01 POINEG	2. 340 2. 0. 0. 0. 0. 0. 0. 0. 0. 0. 0. 0. 0. 0.	102 1.372E+
JCWAK 110 5.0470E-10 EWMM 0. PDTPOS X (JPWAK) * \$5765-01		S N A IRON FLYER S N A IRON FLYER
JEIN JSTAR 291 291 291 284 284 284 284 284 284 284 284 284 284	88 88 50 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	*9023 5.9927E*05 42 *9058 5.9924E*05 43
F 5.51.43E.61 15 5.61.43E.61 15 F 6.440.E 11 K(JFIN) 1.200.E.00 165E-07 SECS, JSTAR	### F## F## F## F## F## F## F## F## F##	828E+10 8-324E+06 7 879E+10 8-941E+06 7
1,7kH 6,5602E=09 EMVNEG 0, 1,1775E 1,2000E+00 0, 1,14N	NACRE DAYCRE DO NACRE	.436E+10 1.176E+10 1. .444E+10 1.220E+10 1.
11ME 0.0319E-07 0.77PuLS 1.2M+1E-06 X(JHNU1) 0.0992E-01	CEYSEC CAYSTON	85940 21947 1 89967 20919 1
CYCLE 2.3721E=05 X(1) X(1) 2.6080E=02 TIME 20		45 43 • 1

TEST 4: FINAL MOMENTUM EDIT AND PORTION OF TIME EDIT

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T 4: FIRST PAGE OF STRESS HISTORIES

MICHOSECS, S

EST 4: STRESS HISTORIES (continued)

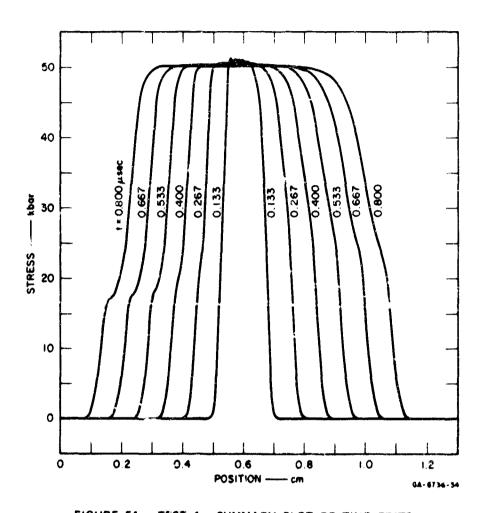


FIGURE 64 TEST 4: SUMMARY PLOT OF TIME EDITS

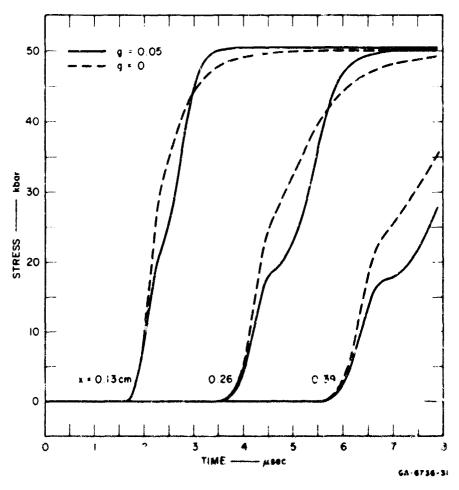


FIGURE 55 TEST 4: SUMMARY OF STRESS HISTORIES AT SIX JEDIT LOCATIONS

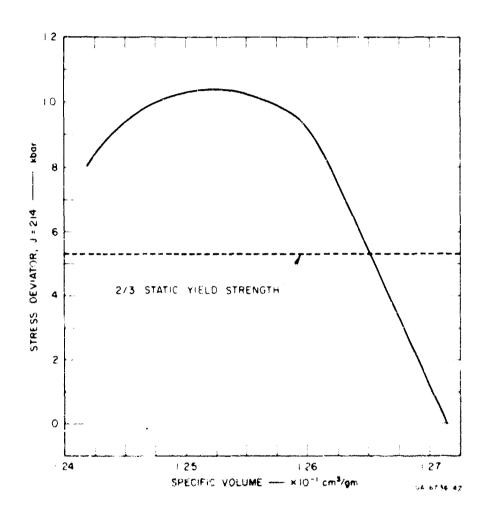


FIGURE 56 TEST 4: SAMPLE PLOT OF STRESS DEVIATOR

# g. Test No. 5

The stress relaxation mechanism of Model 3 (Gilman's model) is tested in Test 5. The configuration is a symmetric impact between two plates of 2024-T8 aluminum. The plates have different relaxation parameters but are otherwise identical. The listing included on the following pages contains the input data, and portions of the initial layout, and final EDIT and stress histories. The EDITs, stress histories, and a deviatoric stress-volume path are shown in the figures.

#### \*\*\*\* SHI PUFF 1 (6400 VERSION) \*\*\*\*

```
DATE #05/13/69
6 NEDTH=
7 STOPS
                  JCYCS =
                                   1000 CKS =
                                                    4.000E+00 TS
                                                                        8.000E-07
8 NATHLS=
                                       1 UZERO =
                                                   2,500E+04
                        RHOS = 7.850E+00 NSRM = 3 NYAM = 1 NPOR = U NCON = 0
IRON FLYEN
8.400E+10 EQSTG #
                                                                        1,600E+00
                                                                         7.2146-12
                                                  1.990E+10 VH #
                                                                        3.2208+05
                                                                        -9.
          5.000E+09 MU = 8.140E+11 YA:
148 CELLS IN 6.000E-01 CM
¥0 =
                               # -1+0E+11 YADD #
NZONES= 1+
                                                   -0.
                                                                        -0.
IRON TARGET
                       RHOS = 7.850E+00 NSRM = 3 NYAM = 1 MPOR =
                                                                      0 NCON # 0
EQSTC = 1.741E+12 EUSTD = 0. EQSTE = 8.400E+10
EQSTM,= 2.500E=01 EUSTS = 0. 1.650E+00
COSQ = 4.000E+00 C1 = 1.500E+01 C2 = 0.
TENS(1) = -1.000E+11 TENS(2) = -1.000E+11 TENS(3) = -1.000E+11
                                       EQSTE = 8.400F+10 EUSTG =
                                                                        1.600E+00
                                                    1.650E+00
                                                                         7.214E-12
          2.000E+03 PHI #
                                         HEE =
                                                   1.980E+10 VM #
                                                                        3.2205.05
                             -0.
8.140E+11 YADD =
*OMAE
          5.000E+00
5.000E+09 MU =
                                                                        -0.
                                                   0.
NZONES= 1.
                148 CELLS IN 6.000E-01 CM
                                                   -0.
                                                                        -0.
```

TEST 5: INPUT DATA

ST 5: PORTION OF COORDINATE LAYOUT

TEST 5: FINAL MOMENTUM EDIT AND PORTION OF TIME EDIT

x(.5942X) 7.6430E-01 PDTXE6	To NOT SO OR NI	2 OR 1/CM	5.000E+00 2.874E+03	3476.5	. 000E+00	.000E+00 2.4	DOGE + DO Z	.000f+00 2.8	600E+00 2-874E-0	.000E+00 2.8	0000	2000-1000 A-44-6000 A-44-600 A-46-600 A-4	0005-00 2-87	000E+00 2.4	.000E+03 2.874E-0	000Eou0 2.874E-8		.002E+03 2-874E-0	.000E-00 2-6	.000E+00	\$ 3000°	• 6	6	0 2.876E-0	0 1	00 2.898E-0	0 2.9536-0	7105.00	JOHN JAMES CONTRACTOR	3.540E-0	0.0 3.746E-9	3-9575-0	1 4,165E-0	1 4.365E-0	*** 10-3E8**	01 4.7292	158E-01 ecovation	3.8 /5E-61 3.8805.443
ESTANCE CONTRACTOR AND CONTRACTOR AN	S 222.538 SECS		S S P 180N FLYE	Z 2	1041 a 2 4	NOW I IS NOW	S N P IRON	₩ . Œ . W .	202	NOW I WON	S Z S IRON	2000 2000 2000 2000 2000 2000 2000 200	7 7 2 4 7 4	2021	S 2 B 280k	S N R IRON	2021 a 2 5	2 2	2027	S N P LEGE		2 : vi (	2002	S N A IRON	S N M IRCH		S R P IRON	S N A IRON FLYE	NOW I W X S	* 2			NO N	S N A 180N FL	S H A SHON FLYE	S & A IRON FLYE	S N & IRON FLTE	43 S N A IRON FLYER
JEIN JSTAR 299 299 291 1808 101972[05] 1808 101972[05] 193 502676[05]	291. CALC TIME S	9 5	516 6.0012E-05	500 6.0004E+05		6-000AE+05	500 6.90042+05	6.0004E-05	500 6.00048.403	STORY OF STREET	6.0004E+05	6.0004E.05	6.0004E+05	00045+03	6-0003E+05	6.0003E+05	6.0003E+05	6,0002E+05	.0002F+03	550 5.44978+05	5.9992E+05	8626 5.9986E+05	8666 5.99782+05	5.9955E+05	8935 5.99426+05	9024 5.9929E+05	9177 5.9967E+05	5.98978+05	5.98886+35	5.38785+05		TOTAL STANDARD STANDS	5.9841F+05	9689 5-98335-05	9739 5.9825E+05	9782 5.9819E+05	9818 5.9814E.05	9858 5.9809E+05
ETOTAL 149E+01 EKWPL 151E+04 Z (JFIN)	-07 SEC3+		7.469E-14	7 21-3640-2 004	101 2.6346-11 7	00 3.671E-10 -	03 6.365E-68 7	7.3205-(7	-04 7.687E-36 7	04 / 363E=03	05 5. 36E-03 7	06 3.738E-02	+06 2.481E-01	3.502€+60	# 25KE+CO	• (c)	A	3 ,E+03 7	1.054E+84	DEGE + 09 3-313E + 04 7-1	2.4426+05 7	5.704E+05 7	1.2035.06 7.	JSOE+04 K. CAURTUR	6.135E+06 7	A.707E-05 7.	1.402.007	1,6812+07 7	.972E+07 7	2.289E+07 7	10 2.6375.07	6 3.007E+07 /	O Sesmoner of	A ALLINEADY 7	10 4-4315-07 7-	10 4.717E+07 7º	*16 4.968E+07 7.	10 5.1886.07 7.
.5557E-09 150 EMVPOS EMVPOS 1.1775E-05 x(.08402) x(.08403)	211. TIME = 8.03371E	Q	018/118	4.800E+00	1.0036.01	5. VerE +01	2. MARF + 02 1.	2.682E+03 &	8.764E+03 1.	2./3/E+04 4.	3-3275-05 3-7	6.3358.05 3.	1.647€+06 2.6	4.092E+06 6,	9. TOSE + B6 1,	Z-200E+0/ 3	9.6276-07 1	1.937E+08 3.	3.6426+08 5.9	6.539E+88 1.0	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	2.407E+09 4.	4.1275.09 6.6	5. (685.09 9.3	9-681E-09 1-	1.1676+10	16 1,348E+10 2,178E+10	1.6556+18 2.	1.798E+16 2	1.9446.10 3.	2.093E+10 3.2	2.244E+30 3.5	2.392E+12 3	2.6566418 A	2-76AF+10 4-	2-86-E+10 4-	0 2-9475+10 4-45	0 3.018E-10 4.
11ME 6.0337E-U7 4.555 DTPULS E 1.8424E-06 0. X.UBND11 X1X 6.1060E-01 1.200	NO. 6 AT N =	<b>5</b>	A CW/SEC DYN/CME	25006 7.794E	25600 2.	25000 1.096E	666F	25000 4.94	25000 1.510E	25000 5.032E	3000 - 000030 3000 - 000030	25000	3 24999	26978 7.	24548	C 54645 3	24904	24930 3.4	E*9 89842	24.765 1.1		24042 4	23539	22969	1 02910	20455 1-93	20321 2	1 19861 2.460/Eviu	18873	18404 3	17928 3.	17451 3	16941 3.7	2 16562 3	٠,	: 4	15296 6	15086 4
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TEST 5: FIRST PAGE OF STRESS HISTORIES

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TEST 5: STRESS HISTORIES (continued)

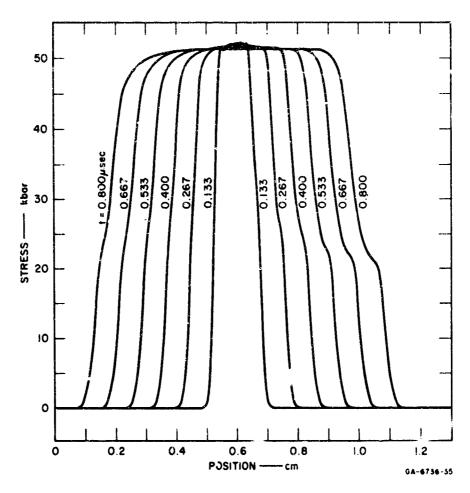


FIGURE 57 TEST 5: SUMMARY PLOT OF TIME EDITS

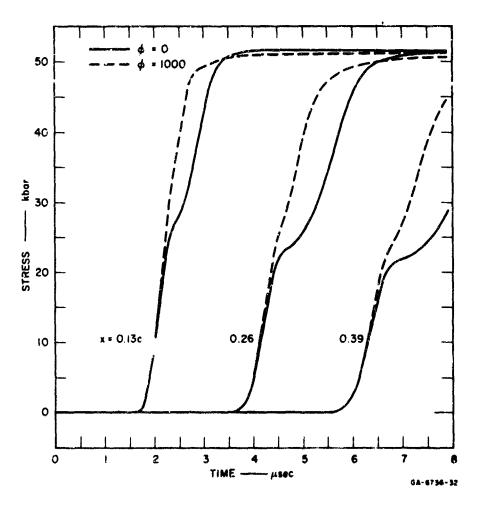


FIGURE 58 TEST 5: SUMMARY OF STRESS HISTORIES AT SIX JEDIT LOCATIONS

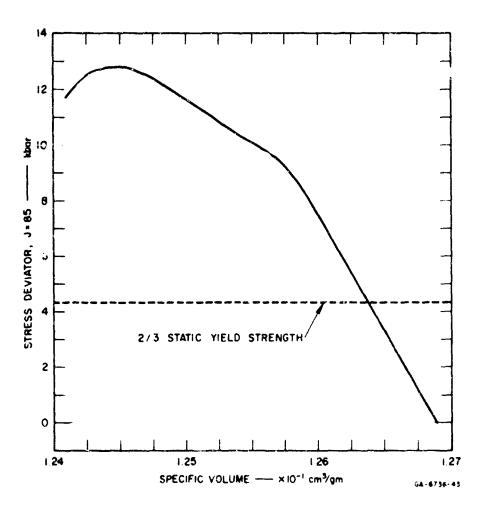


FIGURE 59 TEST 5: SAMPLE PLOT OF STRESS DEVIATOR

## h, Test No. 6

The stress relaxation mechanism of Model 4 (two parameters, variable yield model) is tested in Test 6. The configuration is a symmetric impact between two plates of 2024-T8 aluminum. The plates have different relaxation parameters but are otherwise identical. The listing included on the following pages contains the input data, and portions of the initial layout, and final EDIT and stress histories. The EDITs, stress histories, and a deviatoric stress-volume path are shown in the figures.

## \*\*\* SRI PUFF 1 (6400 YERSION) \*\*\*\*

```
DATE = 04-28/49

IDENT = 6 MODEL 4. TWO-PARAMETER, VARIABLE VIELD RELAXATION MODEL IMPACT OF TWO 2024-TB ALUMINUM PLATES. EACH DESCRIBED BY THE TWO-PARAMETER VIELD MODEL. THE TWO PLATES MAVE DIFFERENT VALUES OF THE VIELD RELAXATION TIME, IV. BUT ALL OTHER PARAMETERS ARE THE SAME.

1 NTGOT= 5 NJEDIT= 6 NREZOM= 0 NSEPRATE 0

2 TEDITS= 5.000E-07 1.000E-06 1.500E-06 2.000E-06 2.500E-06

3 JEDITS= 5.000E-07 1.000E-06 1.500E-06 2.000E-06 2.500E-06

3 JEDITS= 10000 NEDIT= 10000 NPERN = 1 -0

7 STOPS JCYCS = 1250 CKS = 4.000E-06 TS = 2.600E-06

0 MMTRLS= 2 NATFL = 1 UZERO = 2.000E-06 TPLOT = 1

ALFLYER RHOS = 2.705E-00 NRRH = 4 NYAM = 1 MPOR = 6 NCON = 9

EOSTC = 7.550E-11 EQSTD = 1.200E-12 EQSTE = 1.220E-11 EQSTG = 2.040E-00

TEMS(1) = 1.000E-07 TY = 1.000E-01 C2 = 0.

TEMS(1) = 1.000E-07 TY = 1.000E-01 C2 = 0.

NZONES= 1. 100 CELLS IN 2.000E-02 NRRH = 4 NYAM = 1 MPOR = 6 NCON = 0

RESTC = 7.550E-11 EQSTD = 1.200E-01 C2 = 0.

NZONES= 1. 100 CELLS IN 2.000E-07 CBSTC = 1.200E-11 TEMS(2) = -1.000E-11

TRALE RHOS = 2.705E-00 NRRH = 4 NYAM = 1 MPOR = 6 NCON = 0

RESTC = 7.550E-11 EQSTD = 1.200E-12 EQSTE = 1.220E-11 EQDIT = 2.040E-00

AL TARGET RHOS = 2.705E-00 NRRH = 4 NYAM = 1 MPOR = 6 NCON = 0

RESTC = 7.550E-11 EQSTD = 1.200E-12 EQSTE = 1.220E-11 EQDIT = 2.040E-00

T.525E-12

COSO = 0.000E-00 TY = 1.000E-11 TEMS(3) = -1.000E-11

TRALE RHOS = 2.705E-00 NRRH = 4 NYAM = 1 MPOR = 6 NCON = 0

RESTC = 7.550E-11 EQSTD = 1.200E-12 EQSTE = 1.220E-11 EQDIT = 2.040E-00

T.525E-12

TEMS(1) = -1.000E-07 TY = 1.000E-11 TEMS(3) = -1.000E-01

TOTAL RHOS = 2.705E-00 NRRH = 4 NYAM = 1 MPOR = 6 NCON = 0

RESTC = 7.550E-01 EQSTD = 1.200E-11 TEMS(3) = -1.000E-01

T.525E-12

TOTAL RHOS = 2.705E-00 NRRH = 4 NYAM = 1 MPOR = 6 NCON = 0

RESTC = 7.550E-01 EQSTD = 1.200E-11 TEMS(3) = -1.000E-01

T.525E-12

TOTAL RHOS = 2.500E-00 NCON = 0

RESTC = 7.550E-01 EQSTD = 1.500E-00 CN = 0

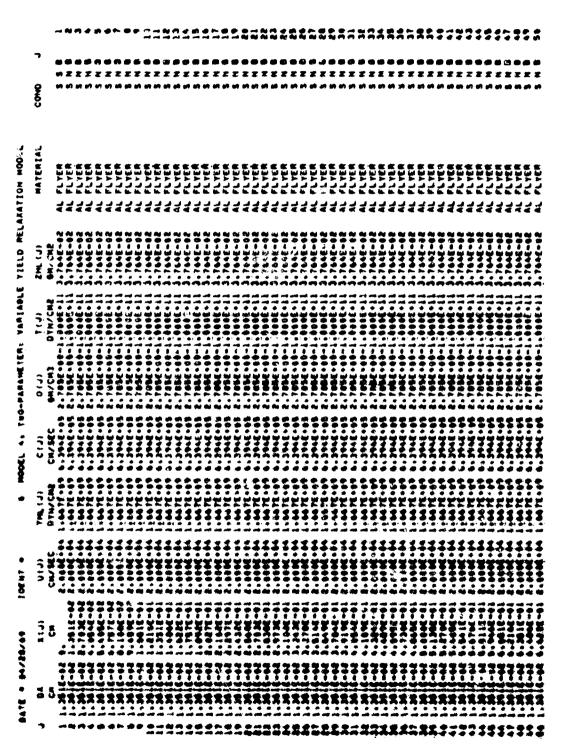
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RESTC = 7.550E-01 EQSTD = 1.500E-01
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TEST 6: INPUT DATA



EST &: PORTION OF COORDINATE LAYOUT

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ETOTAL 2.6442E.01 ENYPL 7.9137E.04 X [1FIN] 4.000E.00		10 2-9695+
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TEST 6: FINAL MOMENTUM EDIT AND PORTION OF TIME EDIT

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TEST 6: FIRST PAGE OF STRESS HISTORIES

| Colored | Colo

TEST 6: STRESS HISTORIES (continued)

X

CUTPUT.

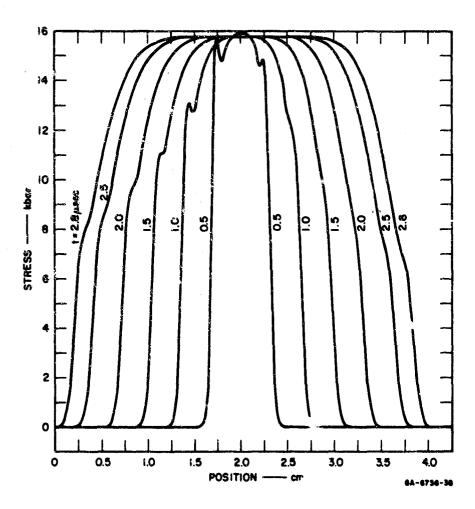


FIGURE 60 TEST 6: SUMMARY PLOT OF TIME EDITS

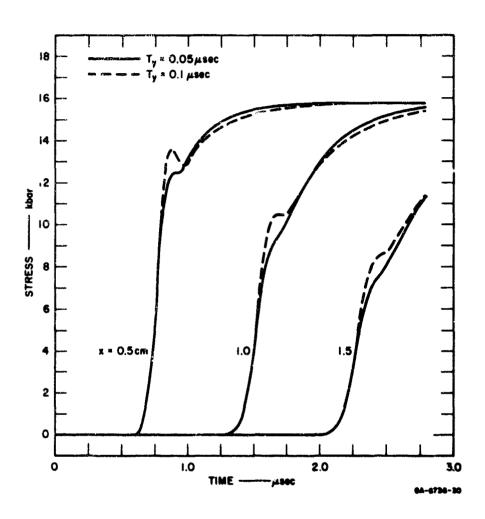


FIGURE 61 TEST 6: SUMMARY OF STRESS HISTORIES AT SIX JEDIT LOCATIONS

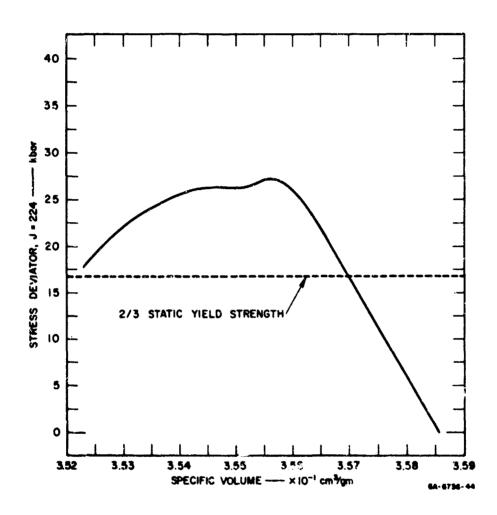


FIGURE 62 TEST 6: SAMPLE PLOT OF STRESS DEVIATOR

## 3. Listing of SRI PUFF 2

In subsequent pages the SRI PUFF 2 program elements for wave propagation calculations are listed. The program requires 51,500 words storage for loading onto a CDC 6400 and 46,500 words for execution. The program and the listings are divided naturally into two segments, the first for computations and the second for plotting. If the plotting cannot be implemented on the user's computer, then the plotting subroutines should be omitted and the call to PUFPLOT should be removed from EDIT. (There is no other link to the plotting routines.) The program elements are included in the following order:

Main program:

SRI TUFF 2

Computational Subprograms:

EDIT

EQST

FMELT

ŒNRAT

HAFSTEP

HYDRO

**JSTRESS** 

POREQST

REZONE

SSCAL

RELAX

BANDRLX

BAUSCHI

Plotting Subroutines:

PUPPLOT

HISTORY

HUGON IO

TPLOTS

```
PROGRAM SRE PUFF (IAPUT, GUTPUT, TAFELC-IAFLT, TAPELE-CUTPUT, TAPEE.
                     TAPET, TAPEE, TAPES, TAPELL, TAPELS, FILPFL=10000)
600000
               . SETS TIME STEP
                               CALLS EDIT. ECTRON. JEZONE AS PECLIFIED
CALLS SCRIBE TO STORE RESILTS AND FOR TERPINAL PRINTCUT
               REAL MATL, PU, MUM, PUP, NEP, NET
INTEGER H. PCRCLS. RIATER, SCLIC. SPALL
MISCELLANEGUS
               COMMON CEF.CKS.DAVG.DELTIM.DCLD.DAMC.CTPIA.DTA.CTAM.DU.DX.EDLC.F.
FIRSY.J.JCYCS.JINIT.JFIN.JSMAX.JSTAR.JTS.LSLB(11).MAXPR(11).A.
NCYCS.NPERN.PQLD.RLAST.SLAST.SPAX.TF.TIPE.TJ.TS.ULAST.UCLC.
                     ALASI, XNGL, XCLC
EQUATION OF STATE - SCLID
               MU(6), MHO(6), RHCS(6), TEAS(6,3), YACC(4), YO(6), JANC(6), NPOR(6,2).
                    A. PUP, NATELS
             EQUATION OF STATE - POPOUS
GURPON AC(6), MUP(6), PORA(4,3), PORB(4,2), PORC(4,3), RHCP(6,3),
1 YACOP(4,3), OREF, RHCFV, PP, NO
 €
               RACIATION CEPOSITION
COMMON $$(300.5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$$100(5),$
C
             IDENTIFIERS
                COPRCE DISCPTIZO, ICENT, JEDITIZO, JREZCHILE, MOATE(3), MEDIT, MEDIT,
              1 MJED11, MR, MREZON, MSEPRAT, ASPALL, ATEDT, ATEX, ATRI 151, TED1T150)
                CCACTIGA INCICATORS

COMMON INFILITER, MIRROR, NORPHAL, PGRCUS, RIATER, SCLID, SPALL

SPALL AND PELAMATION
c
                COENCH MSR(4).TSR(4.4).USP(50).>SP(50).LPLX.A1CX.AEPH.AETH
                CORPOR /PLCT/ IPLCT. IFPLOTIO), LINES, ITPLCT, ATPLOT. IFLAGT. INISTIA),
              1 IFLAT
t
                COMMON ALZSOCI
 100
                CALL SECONCIFICATION & XIN-FIRST
                 RECENS ICE
                 CH-MCVCS-MPERM & ET-MING(G, NTEDT-1) & NT-0 & SF-0.8
                 ************
                                       CALCULATE AND STORE HYCROCYNAPIC CATS
                 CALL PYENG
                            -BIN 8 CALL SELUMDIBERS 8 DELTER-BEAL
PERSONIC BRITS, PRINTS
PERSON, 293 JEG. C) 205,210
                 31ML-216 8
 £
                 IF IPCEIM.
                                                                                                                                                                                             12/17
 205
                 CALTERNAMINE INST
                 UPITELE OPIN-USTAN, TIME, CALTEN, UTS, DINH, SAAP, JSHAD
SPINOSIN, NEDTHI, EQ. CALL EDINCP
SPECOIN, NEDIT). EC. CI CALL ECIT
   210
                                                                                                                                                                                             12/17
 C
                                        STORE DATA IN BUFFER
                 CALL SCRIBE
                        STOP PARAPETERS
                 IF (TIPE .LT. TS) 164,468
IF EM .EQ. JCYCS) 4CC,969
IF (MIJSMAN) .LT. (MS) 126,400
                CALL SSETCH(2.1SETCH)
BC TC (400,500) ISETCH
```

MAIN PROGRAM: SRI PUFF

```
WRITE (18,41)
WRITE (18,40) N.JCYCS,TIME.TS,X(JSPAX),CKS.ISHTCH.CTNH
CALL EDTHON 8 CALL EDIT 8 LSUE(7)=1 8 CALL SCRIBE
PROGRAM FALTS CH CCMPLETICH CF ALL GATA CECKS
466
C
C
          TIME STEP CALCULATION
UTNH-AMINI(SFOOTMIN.AMAXI(1.2001NH..C350SFOOTPIN))
IF (SCURM.EQ. 1.C) 530,504
C
500
C
                   COMPUTATION OF SOURF
         COMPUTATION OF SHORE
SDLEM=1.0
IF(TIME.LT.SSTOPH)515.53C
DO 525 MS=1,ASPEC
IF ((TIME.LT.SSTOP(MS)) .AMC. (TIME.GT.START(AS))) 520,525
SDURN=APIN1(SDURP,SSTCP(AS)-START(AS))
506
515
520
525
          COATIAUE
           CTH-AMINICTHF, C.CI+SOURH)
530
          CH-NCYCS-NPERA
          TIME EDIT

IF (IT) 560,550,535

CALL ECIT SAT=AT+1

IF (AT .EG. ATR(AREZON)) 537,53E

CALL REZONE & AREZON=ARSZCR+1
Ċ
  535
534
          CCATIANE
          IF (NT .EQ. NTEDT) 54C,545
540
545
           11-0
          IF (TIME+CR*OTAH .LT. TEDIT(RT+1)) 540.555
RCYCS=(TEDIT(RT+1)-TIME)/CTRM+1 8 CR*ACYCS
CTRF=(TEDIT(RT+1)-TIPE)/CR 8 IT=)
555
          A-N+1 8 AR-MR+1
1F (DTN+ -GT. 0) 200,545
A-R-1 8 GC TC 400
540
                                                                                                                              12/17
565
         40
```

MAIN PROGRAM: SRI PUFF (concluded)

```
SUBBOUT INE ECIT
         (INCLUDES ENTRIES ECTHOM, ECOUMP, AND SCRIBE)

MAJOR PPINTOLT IS NAMBLED BY THIS ECUTINE

EDIT LISTS COORDIMATE QUANTITIES FOR TIME OF TEDIT

EDITION COMPUTES AND LISTS PEPENTUM CUANTITIES

EDOMMP SIVES A MORE COMPLETE EDIT FOR DEBUGGING PURPESES

SCRIBE STORES SELECTED MESULTS DUBING COMPUTATION AND NAMES
A FINAL LISTING OF STRESS HISTORIES
.....
          REAL MATL, MU, MUP, MUP, AEP, AET
INTEGER M. POROUS, RISTER, SCLIP, SPALL
MI SCELLAKECUS
          COMMON CEF. CXS. CAVE. OBLT IM. DOLD, DR HO. DTP IN . DTM . DU . DX . ECLO.F.
             FIRST, J. JCYCS, JIAIT, JFIA, JSHAX, JSTAR, JTS, LSLB(11, MAXPR(11), N.
             NCYCS, NPERN, POLB, RLAST, SLAST, SMAX, 1F, TIME, 11, 15, ULAST, UDLD,
             MAST. MOD. HOLD
                        EQUATION OF STATE - SOLID
        COMMON COSQLO.4). 1(4.4).C2(4).EMELT(4.5).EGSTA(4).ECSTC(4:.

1 EGSTB(4).EGSTE(6).EGSTG(4).EGST+(4).EGSTN(4).EGSTS(6).NATL(2.6).
             MU(4),RMD(4),RHDS(4),TEMS(4,3),YAOD(4),YO(6),JBAD(4),APGR(4,2),
             H, NUP, MITTELS
                        EQUATION OF STATE - PCECLS
          CORNON AK(6), MUP(6), PORA(4, 3), PCR8(4, 3), FCRC(4, 3), RFOP(6, 3),
             YARGP(4.3. DREF, RHOPY, NP, AC
RABIATICA DEPOSITION
          COMMON SS(300,5),SSTOP(5),START(5),SDLAR,SSTOPP.HSFEC,SSJ,JSS
COGRO LAATE ARRAYS
        COMMON X13001, C13001, CHL13001, D13CC1, DHL13001, EHL13001, H1300, 51.

1 MEN13001, MET13001, P13001, PHL13001, A13001, S13001, S1. (300), T13001, 2 W13001, WHL13001, YHL13001, ZHL130C1

MALFSTEP VALUES
          COMMON SM. ONLAST. DUN. EN. PH. MH. SHLAST, Sh. SPLAST, UN. UPLAST, XH. XMLAST
                         ICENTIFIERS
        COMPON DISCPT(101,10EhT, JEDIT(20), JAEZGh(19), RGATE(3), MEDIT, MEDTH,

NJEDIT, MR.JMEZON, MSEPRAT, MSPALL, MTEDT, hTEX, hTR(15), TEGIT(50)

COMPON INFOLINTEN, MIRROR, MORNAL, POROLS, A INTER, SCLID, SPALL

SPALL AND RELAXATION

COMPON RER(4), TSR(4,4), USP(50), JSP(50), LRLR, MICK, hERM, METH
          COMPON /PLOT/ IPLOT, IPPLOTIA), LINES, ITPLCT, ATPLOT, IPLAST, INISTITY,
             LOLAGI
          COMMON A1 25001
          DIMENSICA [AY(1]), AA(A), NJEB(0], NJB(0), PJE(4)
DIMENSION P1(300), P2(300), S0(300), E(50,50)
          EGNIANTENCE (1988(1)*185)* (1888(5)*185)* (7810(2)*183)
EGNIANTENCE (**8)
ELMENZION ATTACHMENT
ELMENZION ATTACHMENT
ELMENZION ATTACHMENT
                             ATONT FOR EACH EDIT
          1·比例 20mc 1978
           CALL SECTION (CHARGE) & DOM-CHARGE-FIRST
           JETAN C-MEMBEL-JERAN +1,JFIN-1) B APTI-JETANG-JINIT-1
MAITE (18,1020) ATEX,A,TIME,JETAN,CLA
           # (3)-5HL(31-PHL(3)
           11-11017
                         3 3-1
           12-4104 (317MB, 1816 (81)
 ıż
           MAITE (14.1627) : 60 TO 12
           CONTINUE
```

SUBROUTINE EDIT

```
IF (IFPLOT(1) .EQ. 0) RETURN
PI=F-1 5 JEND-JENIT+209
IF (NI .LE. 0) GO TO 35
80 30 MR=1.MI
JEN-JENG(NO)
         10 (MB) = X ( JBM)
15
         MITE (7) MTEX, N, TIME, MPTS, (X(J), &(J), S(J), J-JIATT, JENO), PI,
         (1)0(1),(=1,4)
(4)0(1),(=1,4)
1 18%, 1MC, 4%, 1MJ, 2%, 4MCOMB, 19%, 20MY, AP, ML SD CB AT)
         ENTRY EDOUGE
                                                                                                       ENTRY
c
       WRITE (18,2001) N.J.JSTAR.MSPALL.TIME.(PATL(1,K).MATL(2,K).
1 (COSQ(K,E).[41.61,EC1(K,E).1].1-1.61,CZ(K).EGSTC(K).EGSTD(K).
2 ERSTE(K).ECSTG(K).EGSTD(K).ECSTR(K).ECSTC(K).MJ(K).MOE(K).
3 OMOS(K).VARO(K).VO(K).(EPELT(K.E).1-1 (EMS(K,E).1-1.3).
           AREKS, MUPEL), 4-1, MITAL SS
         MITE (16, 2005)
00 461 #-JINIT,JFM
         SG(E)=X(E+1)-X(E)
         KL-JERT 6 M-1
K3-ARMOLJFEN-L.JONG(A))
MEI WE (18,200) (K.XIR), SO(K), CML(K), EML(K), EML(K), PML(K), SML(K),
         | A(R), T(R), WAL(R), WAL(R), 2M(R), (M(L), 1), 1-1, 3), K-R1, K2)

M | 142 | 46. | F(R-1) | 66 | 70 | 420
         21-#2+1
                  s watte (16-206)) 8 60 TC 412
         P-#+1
         CONTINUE
         PETURA
       PORNAT (11000000000 0-14.40, J-13.00, JSTAN-13.00, ASPALL-13.
1 70. TINE-610.3/(100.2210.40, C-1364.3/46 64%)=12610.3/
2 40 815C-10610.3/)
        2005
       1
         ERTOY ESTRON
MOTPOL-PETREG-O.
JETAND-JETAN-L
                                                                                                       ESTET
5
         POTAL MINESTES CALCULATICS ENVIRON-ENVIRON-ESPACESUM-BRADA-C. SO 200 J-1,257ARC IF (FLJ) .GT . RARE 201,204
         MAN-8(4)
283
          MAA. J
         $11-140-(1)31-(1)10-1-11)
         IF (ENV AT. 0.) 20%.806
ENVOICEMENTS:
245
         68 10 167
EAVES-EAVES-EP
200
         CART LANG
```

SUBROUTINE EDIT (continued)

```
C
                          CALCULATE HOMENTUM OF MAIR PLISE
           ERYPL=0.
          EMVPL=EMVPL+U(JM)+.5+C(JM)+(X(JP+1)-X(JP-1))
209
          EMPPL=EMPPL=U(JM)=.5*E(JM)=1

IF (JM .LT. JSMAX 210-211

IF (UJM-1) .GT. C.) 211-212

IF (JM .GT. 1) 212-213

JM=JM-1

GO YO 209

JM=JSMAX+4
210
212
213
C
                          CALCULATE MEMENTUM OF PRECUESOR
           FHVPR:30.
          ENVPR-ENVPRO-SOULINGO (JP) +(X(JP+1)-X(JF-1))
214
           SF (JM .GT. JSTAR) 230,215
JHUML
215
           6C TC 214
           EMYPP=EMYPL+EMYPR
216
          DTPP-ENVPP/SMAX
DTPULS-ENVPL/SPAX
E
                          CALCULATE PAXIPUR POTENTIAL POPENTUR OF VAPOR
           EN VBM=0 .
           OC 220 J=1,JSTAR

IF (J .GT. JBN0(R)) P=R+1

DE=EHL(J)=ECSTE(R)
           IF(DE .GT. 0.) 219,221
UAVG=(SGRT(U(J-1)*U(J-1)+2.*DE)*SGRT(U(J)*L(J)+2.*DE))/2.
219
           (L) JIX+DVAU+MBVM3-MBVM3
220
           CONTINUE
                          PRINT OUTPUT VARIABLES
         GETHEN
          FORMATIZSHI MCPERTUP EDIT/)
         FORMAT (/-, 7x, 5HCYCLE, 8x, 4HTIME, Ex, 4HDTNH, 9x, 3HJTS, 6x, 6HETCTAL, 8x, 4
HJFIN, 7x, 5HJSTAR, 7x, 5HJSPAX, 8x, 4HSPAX, 4x, 8HX(JSMAX), /, 712, 2E12, 4, 1
212, E12, 4, 3112, 2E12, 4, /, 8x, 4HDTPP, 6x, 4HDTPLLS, 6x, 6HEPVMEG, 6x, 6HENVP
30S, 7x, 5HENVPL, 7x, 5HEPVPR, 7x, 5HEPVPP, 7x, 5HENV8M, 6x, 6+PCTPOS, 6x, 6HPD
4THEG, /, 10E12, 4, /, 8x, 4HX(1), 4x, 8Hx(JBND1), 3), 8HX(JBND2), 4x, 6HX(JBND
531, 5X, 7XX(JFIN), 7x, 5HJRNAX, 8X, 4HRMAX, 4x, 8HX(JRPAX), 7, 5E12, 4, 112, 2E
          412-41
           ENTRY SCREEK
                                                                                                                             ENTRY
Ç
          ## (M .EG. 1) 250,291
18UF-G. 8 MOCLUS-29
00 286 [=1,hubdit
00 2Fi IM-1,hmtrls
1F (JEDIT(I) .LE. JBND(IMI) 286,284
250
284
284
           CONT INUE
          RIECE)=IN
 251
                                                                                                 #418+171=ATS
220
```

SUBROUTINE EDIT (continued)

```
r-3.1*((36)-(36)*)-(31)$
            ALIBH-(3412)-P(12)191.E-R

DO 223 1=1.6

LE-JED17(1) & IP-18+1 & ALIBH-341419-1.E-9 & ALIBM-11E,33

ACIBH-1910R(JE)+3.E-9 & ALIBM-27)-ALIBM-391-75(JE)-P(JE)10-1.E-9

IF (D(JE) .6Y. O.) ALIGH-21)-1./O(JE)

MARHJELLI & RSRPJI-MSR(MJE)-1

GU TO (225 .2243,2241,2241,2242,2242) RSRMJ1

ALIBM-331-MER(JE) & ALIBM-391-RET(JE) & GG TO 225

ALIBM-331-MER(JE) & GO TO 225

ALIBM-331-MHLJE)+1.E-9
2242
2243
             CONTINUE
              I MIFO INUFOR
             IF (1607 .60. MODLUS) 233,246
IF (UNIT.6) 233,239,500.501
SJFFER QUT (6,1) (A(1),A(50**EDLLS))
233
235
             63 TO 270
63 TO 270
1F (18UF .EG. 2*MCCLUS) 342,270
IF (UMIT.6) 242.249.500.501
SUFFER OUT (4,1) (A(50*MODLUS*1),A(100*PODLUS*))
240
 242
245
270
             RETURN
            290
                                                                                                                                                             12/17
  351
352
335
 340
 342
345
                     ZERO CCHRCA
             CC 301 [=1,2500
361
             A( | ) = Q .
                      SECOND(XSTART)
             IF (IPLCT .EC. 0) 60 TO 307
                                WRITE PLCT FEADING DATA IN TAPE 7
             OG 302 I=1.6
              JE = JEDI 1(1)
302
             XJEC(1)=X(JE)
            XJEC(I)=X(JE)
WAITE (1) N,NJEDIT,(JEDIT(I)=1-6)-(XJEC(I),1=1-6)

DUFFER IN THE ARRAY STORED ON TAPE 6

INDX=0 s ISUF=MODILS s NBUF=(N-1)/PCGLUS+1 s LENGTH=MGGLUS
BUFFER IN (6-1) (A(1),A(500MODILS))
ED 325 NB=1,NBUF
II=1+RODIUS=PCO(AB+1,2)
IF (UMIT,6) 308,3C5,5C0,5C1
IF (NR -EQ. ABUF) 310,311
367
309
369
             RESET LEAGTH FOR INCOPPLETE CLCCK
IF 1MOD (N.MCDLUS) .EQ. 0) GO TO 314
LENGTH-MOD(N.MODLUS) 8 GG TC 314
310
            IF 111 .E4- 1) GG TG 313
BUFFER IN 16-1) (A(1),A(5U+ROOLLS))
GO TO 315
BUFFER IN 16-1) (A(5C+RCOLUS+1),A(1CC+FCELLS))
111
312
313
             IF (II .EQ. 1) WRITE (IN. 1041)(.2017(JE), JE>1,00
115
                               PRINT OUT OF FIRST SCRIBE FISTGRIES
             DG 320 1-11.12
CG 320 J-6.11
JM(J,1)-61J+6.11
           MRTE 118:10+001(18(J+17:4=1:5):(8(J+1):J+(J+1):J+6:1k):

1 (8(J+1):J+17:2G):J+81:E2:

PREPARE TAPE 19 FCR SECCEC SCRIEE

IF (LMLX .GT. 2) MRTE (16) ((3(J+1):J+1:2):(8(J+1):J+19:3G):J+12:
```

SUBROUTINE EDIT (continued)

```
PREPARE TAPES FOR PLCTTING

IF (ISPLOT(2)) ISPLOT(3) .RE. 0) MRITE (7)((8(J.1),J=3,11),I=11,12)

IF (ISPLOT(3)) ISPLOT(4) (ISPLOT(2) .NE. C) RAITE (8) ((8(J.1),1))
            J=27,32); I=[1:125

IF (IFPLOT(4) .NE. C) WRITE (9) ((B(J:1),J=21,26),I=[1:12)

IF (IFPLOT(5) .NE. G) WRITE (11) ((B(J:1),J=33,30),5=[3,12)
125
            CONTINUE
            REWIND 19
            PRINTGUT OF STRESS HISTORIES (SECOND SCRIBE)

IF 4LRLX .EQ. 11 6QC, 330

RP=1H-11/50+L 1 MD=50
             12-5C
            00 345 IP-1-NP
            FRINT 1091. (JECIT(JE),JE=1,6)
SF (IP .EQ. NP) 332,34C
JF (MOD(N,50) .EL. 0) 340,336
333
            52-POC(A, 50)
13-# INO(12, 25)
334
340
            343
            CONTINUE
345
600
            CALL SECOND (XEAD)
            MELL SECURCIATION S REWING E A REMIND 9 3 REWIND 12

CALL TO PLOTTING PRESPAK

IF (IPLOT ... EQ. 0. "3 TO 620

PRINT 450
C
             CALL PUPPLOT INTEX, SEENTS
            THEN TO BE REMIND & S. REBIND 9 S. REWIND 11
CALL SECOND (FINAL) S. CUR4-FINAL-FIRST
CUR-MENS-MENS-MENT S. DUR2-MEND-FIRST S. CLR3-FINAL-MEND
MRITE (18,5012) DUR, DURZ, DUR3, CLR4-
            STATUS CHECK PRINTS
IF (NO .Eq. nbuf) Length=Rod(A,PCCLUS)
 500
          MRITE (18-1061) NB.LENGTH S GC TG 600
IF (NB .EQ. NBUF) LENGTH-MODIN, MODILS)
MRITE (18-1062) NB.LENGTH S GC TC 600
FORMAT (13H PUTPLCT CALLED //)
FORMAT (15-0,4F10.1,6F5.3,R1),F5.0,F7.3,F0.3)
FORMAT 101ML SCRIBE CUTPUT, TIME IN MICROSECS, S IN KBARS, DTHM IN
1 NAMOSECS, DELTIM IN SECS //
2 SIM N TIME S12/SCJ1 S23/SCJ2 S34/SDJ3 S(IS,
3 7H) S(I3,7H) S(I3,7H) S(I3,7H) S(I3,7H) S(I3,7H)
4 2N) JTS GTAM DELTIM)
FORMAT(27H EQF ON UNIT 6 IN SCRIBE. 5H NL=12, 9H LENGTH=13)
FORMAT(215,F10.3.15)
             MAITE (18,1042) MB.LENGTH & GC TG 600
 501
 498
 1041
 1041
 1070
             PORMAT(215, F1C.3, 15)
 1080
           FORPAT (P5.0,P10.3,12E10.3)
PORMAT (17MOTINE IN SCRIBE =F10.3/17H CCPPUTING TIME =F10.3/
1 17M PLOTTING TIME #F10.3/17H TOYAL TIME =F1C.3)
```

SUBROUTINE EDIT (concluded)

```
SUBAGUTINE EGSTIEL.DJ.PJ.NJ.CJ)
                          COPPUTES PRESSURE AND SOUND SPEED FOR SCLIDS
                                                  MIE-GRUNEISEN FER CEPFRESSIEN
EXPANSION EQUATION OF STATE FOR DENSITIES LESS THAN RHGS
                          REAL PATL, NU, KLP, FUF
RISCELLANEGUS
                          COMMON CEF.CXS.DAYG.CELT IN.CCLL.CAMC.ETP IN.UTR.ETNF.DU.DX.EOLD.FC
FIRST.J.JCYCS.JIMIT.JFIN.JSMAXJJSTAR.JTS.LSUB(11).MAXPR(11).N.
XCYCS.MPERN.POLD.PLAST.SLAST.SMAX.FF.TIME.TJ.YS.ULAST.UGLD.
                                   NLAST, XNOW, XOLE
EQUATION OF STATE - SCLID
                          CRMMON COSG(6.6).Cl(6.6).C2(6).EMELT(6.5).EGSTA(6).EQSTC(6).

EGSTB(6).EGSTE(6).ECSTB(6).ECSTM(6).ECSTA(C).EGSTS(6).MATL(2.6).

ML(6).MD(6).RHCS(6).RHCS(6.2).YACC(6).WD(6).BD(6).RDC(6).
                                                                EQUATION OF STATE - PERCUS
ε
                          CORRON AK(6) MUP(6) PORA(4,1) PORB(4,2) PORC(4,3) RMCP(6,3) RACP(4,3) OREF, RMCPV, MF, NC
C
                          TSZ-ECS (H(MJ)+GMMU & PJ-(EJ-TS1)+DJ+TSZ
                          TS2=ECS(H(MJ)+GNMU & PJ=(EJ-TS1)+0J+TS2

IF (CJ .ME. 0.) 9-2C

CSG=(EJ-TS1)+0(TS2+.5+GMMU)+TS2+(EQSTE(RJ)-TS1)+EQSTA(RJ)+(2.+VJ-1.

| )+VJ(PJ+TS2+OJ + 2.+MVM/1DJ+DCLD)

IF (CSC .LT. 0.) GO TO 30

CQ=CSQ+CJ+CJ & CJ=CSQ+CJ/CQ+.25+CQ/CJ & GC TC 2C

EQST FCR CCMPRESSEC 20KES

PM=(!EQSTS(MJ)+EMU+EQSTD(MJ)+ERU+EQSTC(PJ)+EPU
G=EQST(MJ)+EMU+EQSTD(MJ)+ERU+EQSTC(PJ)+EPU
G=EQST(MJ)+EMU+EQSTD(MJ)+EMU+EQSTC(MJ)+EPU
G=EQST(MJ)+EMU+EQSTD(MJ)+EMU+EQSTC(MJ)+EPU
G=EQST(MJ)+EMU+EQSTD(MJ)+EMU+EQSTC(MJ)+EPU
                                                                                                                                                                                                                                                                                                                            12/17
 10
                         11
                                                                                                                                                                                                                                                                                                                           12/17
 20
                          LOUISTINGTON TO THE CONTROL OF THE C
 25
 110
```

SUBROUTINE EQST (complete)

```
FUNCTION FRELT(EJ.FJ)
C
          COMPUTES VARIATION OF STRENGTH OR POCULUS WITH INTERNAL ENERGY
          REAL MATL.MU.MUM
        MISCELLANEOUS

COMMON CEF.CKS.DAVG.DELTIM.DOLD.DRHC.DTFIN.DTN.DTNH.DU.DX.EGLD.F.

FIRST.J.JCYCS.JINIT.JFIN.JSMAX.JSTAR.JTS.LSLB(11).MAXPR(11).N.

MCYCS.MPERN.PGLD.RLAST.SLAST.SMAX.TF.TIME.TJ.TS.ULAST.UCLD.
C
        3 XLAST, XNGB, XCLD
        EQUATION OF STATE - SCLID

COMMON COSQ(6,6),C1(6,6),C2(6),EMELT(6,5),EQSTA(6),EQSTC(6),

1 ECSTD(6),CQSTE(6),EQSTG(6),EQSTM(6),EQSTM(6),EQSTS(6),MATL(2,6),
C
            MU(4), RHO(6), RHOS(6), TEAS(6, 3), YACC(6), YO(6), JBNO(6), NPOR(6,2),
        3 MANUN, ANTRES
C
         IF (EJ .GE. ERELT(MJ.1)) GC TC 3C IF (EJ .GT. ENELT(MJ.2)) GC TC 1C 2C IF (EJ .GT. O.) GG TC 1O FOR NEGATIVE ENERGY
C
         FRELT =1.
         RETURN
                       FOR ENERGY IN FIRST REGION
         RR=EJ/EMELT(MJ.2)
         FMELT=1.-RR+(1.-EMELT(MJ,4)+4.+EPELT(PJ,3)+(RR-1.))
         RETURN
         FOR ENERGY IN SECOND REGION

OR=(EMELT(MJ,1)-EJ)/(EMELT(MJ,1)-EMELT(PJ,2))
20
         FMELT-RR+(EMELT(HJ,4)-4.+EMELT(MJ,5)+(RR-1.))
         RETURN
C
30
                       FOR ENERGY ABOVE PELTING
         FMELT-0.
         RETLAN
         END
```

**FUNCTION FMELT (complete)** 

```
SLBROLTINE "FARAT
          REACS INPUT DATA AND INITIALIZES ARRAYS ** READS INPUT CARDS
                    READS INPUT CARDS
COMPRITES COORDINATE LAYOUT
CCMPUTES RADIATION DEPOSITION FOR A RADIATION PROBLEP
INITIALIZES DENSITY, EMERGY, YIELD, SCUAG SPEED, TEASILE STRENGTH,
COMDITION INDICATORS, PARTICLE VELOCITY
PAINTS INITIAL LAYOUT
           REAL MATL, MU, MUN, MUP, NEM, MET
           INTEGER H. PCRCLS, RIATER, SOLIC, SFALL
           INTEGER HH
C
                         MISCELLANECUS
          COMMON CEF, CRS, DAVG, CELTIM, COLG, CRMC, CTPIM, CTM, DTMM, DU, DX, ECLD, F, FIRST, J, JCYCS, JIMIT, JFIM, JSMAX, JSTAM, JTS, LSLB(11), MAXPR(11), A, MCYCS, NPERA, PCLD, RLAST, SLAST, SMAX, TF, TIME, TJ, TS, LLAST, UOLD,
              XLAST.XNCW.XQLC
          XLASI,XNCH,XULC

EQLATICN CF STATE — SCLIC

COMMCN COSG(6,6),C1(6,6),C2(4),EMELT(6,5),EGSTA(6),EGSTG(6),

L EGSTG(6),EGSTE(6),EGSTG(6),EGSTH(6),EGSTS(6),MATL(2,6),

Z MU[6),RHO[6),RHOS(6),TENS(6,2),YAGD(6),YC(6),JBNO(6),HPGR(6,2),
              M, MLM, NM TRL S
          EQUATION OF STATE - POROUS
COMMON AK(6),MUP(6),PORA(4,3),PCRB(4,3),PCRC(4,3),RFC?(6,3),
L YADDP(4,3),DREF,RFGPY,MP,NC
          RADIATION DEPOSETION
COMMON SS(300.5),SSTCP(5),START(5),SCLRM,SSTOPM,NSPEC,SSJ,JSS
COORDINATE ARRAYS
C
C
         COMMON X(300),C(300),CHL(300),C(300),CHL(30C),E^L(300),H(300,3),
1 NEH(300),ET(300),PE(300),PE((300),H(3CC),S(3CG),SFL(300),T(300),
2 U(3CC),UHL(3CO),YHL(300),ZHL(300)
HALFSTEP VALUES
           COMMON CH, DHLAST, CUH, EH, PH, RH, RHLAST, SH, SHLAST, UH, UHLAST, XH, AHLAST
C
                          IDENT IF TERS
           COMMON DISCPT(10), IDERT, JEDIT(20), JREZCA(15), ADATE(3), AEDIT, MECTA.
         1 NJEDIT, NR, MREZON, NSEPRAT, NSPALL, NTEDT, NTEX, NTR(15), TEDIT(50)
                          CONDITION INDICATORS
          COMMON INF-LIATER, PIRROR, NCAPAL, FCPCUS, PINTER, SOLIC, SPALL SPALL AND RELAXATION
C
           COMPON NSR(6).TSR(6.6).USP(50).XSP(50).LFLX.NICK.NEPH.NETH
C
           COMMON /PLOT/ IPLOT.IPPLOT(6).LINES.ITFLCT.ATFLCT.IFLAGT.IFIST(9).
              IFLAGI
C
           COMPON A(2000)
          DIMERSIEN AC(109),AA(4,4,15),B(6,4,15),EDGE(4,4,15),EI(109'
| RHOC(4,4),TBL(109),ACCA(4),ACE(4,4)
| DIMERSIEN DELFIN(4,5),CELX(4,5),TH(4,5),ACELLS(4,5),MZOAES(4),
         1
             1EQLEV(10).RLX(6.8)
           EQUIVALENCE (AC.( .:AA.C(11C)),(8.0(5G)),(E.GE.DHL(29G)),
(EI,EHL(23G)),(HHCC,H(39)),(TEL,H(75)),(NGCN,H(104)),
(MDE,H(19G)),(DELFIN,H(226)),(DEL),H(256)),(TH,H(206)),
              INCELLS.H(316)).INZCNES.H(346)).(CEF. JECUIV)
ε
           REGIA=FIRST
           CALL NEMSET(0., TEQUIV, 9770)
FIRST-EFSIN
100
           LINTER-IAL & NORMAL-IAN & PGRCLS-IAF & PIRRCA-IAF
AINTER-IRA & SCLIG-IAS & SPALL-IAS & NORM-SHÆRNO
                                                                                        PIRRCH-18P
           LELX-1
               READ AND PRINT DATA
AD 45,A1,ICENT,(DISCPT(1),1=5,10)
C
           READ
           CHECK FOR ENG OF LAST CATA CECK
IF (EC.).101 102,103
           STOP
 102
            MITE (10,31)
           CALL SCATE CATS
           MITE (18,43) DAT
MITE (18,43) A. IGENT, (DISCPT(1), 1-5,10)
                                       DATE - 8 DISCPT(Z)=UAT
IDENT - 8 ENCODE (16.33.01SCPT(4)) IDENT
           DISCPY(1)-10+
                                      DATE -
```

SUBROUTINE GENRAT

```
59.IND.(A(I).I=1,8)
EC. LCH ) GC TO 1C8
105
                   REAC
                   IF (IND .EC. 1CH )
WRITE (18.59) IND, (A(1), (-1, 6)
GO TC 1C5
                   EACKSPACE 10
REAC 74,A1,A2,NTEDT,A3,A4,NJEDIT,A5,A6,NREZCN,A7,A8,NSEFRAT
MRITEIL8,74;A1,A2,NTECT,A3,A4,NJECIT,A5,A6,NREZCN,A7,A8,NSEFRAT
IF (NTECT -EQ- C) GC TC 12C
NT-MINC(7,NTECT)
108
                 IF (NTECT -EG. C) GC TC 12C

NT-MINC(17,NTEDT)

READ 76,A1,A2,(TEDIT(1),I=1,NT)

IF (NTEDT -LE- 7) GO TO 12C

READ 79,(TEDIT(1),I=2,NTEDT)

WRITE (18,7)(TEDIT(1),I=2,NTEDT)

WRITE (18,7)(TEDIT(1),I=0,NTECT)

IF (NJECIT -EG. C) GC TO 125

AJ-NINO(14-NJEDIT)

REAC 78,A1,A2,(JEDIT(1),I=1,NJ)

WRITE (18,70)A1,A2,(JEDIT(1),I=1,NJ)

IF (NJECIT -LE- 14) GO TO 125

READ 8C,(JECIT(1),I=15,NJECIT)

WRITE (18,80)(JECIT(1),I=15,NJECIT)

WRITE (18,80)(JECIT(1),I=15,NJECIT)

WRITE(18,70) A1,A2,(NTR(1),I=1,NREZCN)

WRITE(18,70) A1,A2,(NTR(1),I=1,NREZCN)

WRITE(18,70) A1,A2,(JREZCN(1),I=1,NREZCN)

WRITE(18,70) A1,A2,(MTR(1),I=1,NREZCN)

WRITE(18,70) A1,A2,(MTR(1),I=1,NI)

WRITE
125
130
                    IF (MAXPRN .EC. 0) GC TC 135
READ 78,A1,A2,(MAXPR(1),I=1,11)
batte (18,78)A1,A2,(PAXPR(1),I=1,11)
                     READ 72.A1,A2,U5VC5.A4,CK5.A5.TS

MRITE (10.72)A1,A2,A3,U5VC5.A4,CK5.A5.TS

T3,A1,A2,A3,U5VC5.A4,CK5.A2,U2ERG.A4,IFLCT

T3,A1,ANTRLS.A2,ATRL.A2,U2ERG.A4,IFLCT
135
                    READ
                     WRITE (18.73)A1, MATRLS, A2, MATFL, A3, UZERC, A4, IFLCT
                                     **** 4331-4 ****
                    DO 29C P-L-AFTRLS
                    WRITE (18.69)
REAC 90, MATL(1, 2), MATL(2, P), AL, RFCS(P), A2, NSR(M), A3, NYAM,
                    MRTTEELR, GOTHATLEL, MI, MATLEZ, MI, AL, MHCSEPI, AZ, ASREPI, AZ, NYAP,
                          A4.NPGREH.LI.A. . ACCAEM)
                   RHO(M)=RHOS(M)
                                                           READ IN EOST YARTABLES
                                                                  Al.AZ, ECSTCIM), A3, A4, EQST: (M), A5, A6, ECSTE (M), A7, A6,
                 LEGSTG(N).A9.A10.EGSTH(N).A11.A12.EGSTS(P).A33.A14.EGSTN(N).A15,A16
EGSTA(N)-EGSTG(P) **RCS(P)/EGSTE(P)
EGSTN(N)-EGSTC(N)/(EGSTG(P)*EGSTE(P)**RCS(P);
                 MRTE (18-92) AL-AZ-EGSTC(M), A3, A4, EGSTC(M), A3, C4, EGSTE(M), A7, A8, 

BEOSTC(M), A9, A10, EGST+(M), A11, A12, EGSTS(M), A13, A14, EGSTE(M), A15,
                 ZALG.ECSTAIR)
                                     *** REAG IN VISCOSITY AND TENSILE VARIABLES ****
                    READ
                                                     42, A1, A2, CCSQ(M, 41, A3, A4, C1 (P, 4), A5, A6, C2 (P)
                    haite tie, 43] al.a2.ccs(ip.o), a3.a4.cl(p.e), a5.a4.c2(n)

REAC 92.a1.a2.tems(m.l), a2.a4.tems(p.2).a5.a6.tems(p.3)

MATTE(18.92) A1.a2.tems(p.l), a3.a4.tems(m.2).a5.ae.tems(m.3)
                                                           READ IN STRESS MELANATICA VARIABLES
                                                  CERRESPENCENCE OF REX DITH MEDEL AND CEPPER VARIABLES
                                     CEMION TSR(1)
                                                              ALX(1)
                                                                                          121
                                                                                                               (3)
                                                                                                                                                       151
                                                                                                                                                                           (4)
                                                                                                                                                                                               (7)
                                                                                                                                                                                                                  (0)
                                   AMELAS. TRLE
                                                                                          12
                                                                                                              336
                                                                                                                                   44
                                                                                                                                                      GEE
                                                                                                                                                                          EFS
                                                                                                                                                                                               1044
                                                                                                                                                                                                                  ATO
                                     GILPAR
                                                               CEE
                                                                                                              eet
                                                                                                                                  YP
                                                                                                                                                       PARC
                                                                                          11
                                     2-P Y.
                                                               TRLX
                                     BALSCH. XPG
                                                                                                           PUUA
                                                                                      X.P
```

SUBROUTINE GENRAT (continued)

```
C
                         MSRM1=MSR (P)+1
                        NSRMI=NSK(P)+1
GC TC (220,210,212,212,210,210,210, NSRM)
REAC 94,(A(I),RLX(M,I),I=1,4)
WRITE (18,94)(A(I),RLX(M,I),I=1,4)
RLX(M,5)=RLX(M,6)=C. S GC TC 214
READ 94, (A(I),RLX(M,I),I=1,8)
WRITE (18,94) (A(I),RLX(M,I),I=1,8)
210
212
                        IRLX=2
EG 216 IR=1,6
214
216
                          TSR(M, IR)=RLX(M, IR)
                        IF (NSR(N) .EC. 5) TSR(N,3)=1.2230TSR(N,2)
EMGLT(M,1)=EMGLT(M,2)=1.E20 8 EPELT(P,4)=1.
220
                         IF (NYAP .EC. 0) 60 TO 240 IF (NYAN .EQ. 2) 60 TO 23C
C C C
                                                                READ IN VIELD VARIABLES ****
                        REAC 92,41,22,400, EA, 40,61,50, 92,41,420, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10,100, 10
                         YO(M)=YCS
                         IF (MSR(M) .hE. 5) YADD(M)=.6666670YACC(P)/(RFC3(M)+1.2-.50YCS/
                            MU(M);;
                        MU(M)=1.3333339PL(P)
IF (NYAP .EC. 1) 60 TO 24C
                                                                    READ IN MELT VARIABLES
230
                                                                76.A1.A2.(EPELT (P. I), I=1.5)
                        WRITE (18,76)A1,A2, (EMELTIM, 11,1=1,5)
                                                                    REAC IN POROUS VARIABLES
                        IF (MPCR(M,1) -EC. 0) GO TO 25G
NP=NPOR(M,1) S PP=RP+1 S APCR(P,2)=PF
READ 76.A1.A2.(%PCF(I,MP),(=1,6)
NRITE (18,76)A1.A2.(RPOP(I,MP),I=1,6)
                        RHO(R)=RHOP(1,PP)
READ 77,A1,A2,(CCSQ(H,12,1=1,5)
bRITE (10,77)A1,A2,(COSC(P,1),1=1,5)
PEAC 77,A1,A2,(CI(H,1),1',5)
URITE (10,77)A1,A2,(CI(H,1),1',5)
URITE (10,77)A1,A2,(CI(H,1),1',5)
                        CG 245 A=1,AP
PGRA(A,PP)=P1=P2
                        PGRB(R-MP)=(P2-P1)-AZ-PZ-A3-A4-DELP-A5-A6-YACOP(R-MP)

NRTTE (18-92)A1-A2-PZ-A3-A4-CELF-A5-A6-YACCP(R-AP)

PGRB(R-MP)=(P2-P1)-MPCP(R+1-MP)/(RMCP(R+1-MP)-AMCP(R-MP))
                        PORCININP)=4.=DELP=RHOP(R+1,MP)=RHCP(A,PF)/(RHCP(N+1,MP)
                               -RHCP(A,#F))##2
                        COMTINUE
                        COMINUE

COMINUE
                        PERBIG-PRIPE & PERLINFATIONELY
REAC 92,41,42,48(F),43,44, RLP(F),45,46,70(F)
MITE (18,92)A1,42,48(F),43,44, PUFFP),45,46,70(R)
                        REAL
                        PUPEPIOT. 3333330NUF(R)
                                            COOP REAC IN EDGE VARIABLES
                        IF ENCCHEM) .EC. O1 GO TO 240
                        MCONST-MCON(N)
                       DO 200 AC-1, MCCHSY
REAC 94, A1, MOE (#, MC ) . A2, A3, RHCC (P, RC)
                      DO 260 AC-1, MCCHST

REAC 34.A1, MOCEM, MC1.A2, A3, RHCC(P.AC)

MB1TE (18, 54.4A1, MCE(P.AC), A2, A3, RHCC(P.AC)

MDEC-MOCEM, MC1 8 MOE1-MOGO-1

MDONHIROTY, AGED 8 AC1-MINOTY, ACE11

READ PW, A1.A2, (EDGE(M.MC, AC), AD-1, ACD)

ARTTE (18, 74.1A1, A2, (EDGE(M.AC, AC), AC-1, ACC)

IF (ACC) 14, 47. GC TO 232

MEAC 34, (EDGE(M.AC, AD), MO-2, ACED)
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```
MRITE (18,79)(EDGE(F.MC.AD).AE=E.ACEC)

BEAD 76,AI.AZ,(AA(M.NC.MD).ND=1.ND1)

MRITE (18,76)AI.AZ,(AA(M.NC.MD).AD=I.AD1)

IF (NDE1 -LE. 7) GC TC 256

REAC 79.(AA(M.NC.ND).AD=E.KCE1)

MRITE (18,79)(AA(F.AC.AD).AD=8.ACE1;

BEAC 76,AI.AZ.(BEM.NC.ND).AD=1.AD1)

ARITE (18,74)AI.AZ.(BEM.AC.AD).AC=1.NC1)

EF (ACF1 -1F - 2) CC TC 2AD
253
256
         IF (ACE1 -LE. 7) GO TO 240
REAC 75,(8(M,MC,AD),AD=8,ACE1)
MRITE (18,79)(8(P,AC,AD),AD=8,ACE1)
260
          COATINUE
                          READ IN ZONING VARIABLES
                         62.41.AZCAES(M).A2.NCELLS(M.11.A3.A4.TH(M.1).A5.A6.DELX
         READ
200
        1 (M. 1).A7,A8,DELFIN(M.1)
          hrite (18,62)A1, RZCRES(N), A2, RCELLS(P, 1), A3, A4, TH(N, 1), A5, A6, DELX (M, 1), A7, A8, DELF IR(N, 1)
          IF (NZGNES(N) -1
AZCN-AZCNES(P)
ED 285 N=2.NZON
              (NZGMES(M) .EC. 1) GC TC 290
        READ 64.A1.A2.ACELLS(H.N), A3.A4.TF(P.N).A5.A6.DELX(H.N).
1 A7.A8.DELFIN(H.N)
bRITE (18.64)A1.A2.ACELLS(P.N).A3.A4.TF(P.N).A5.A6.CELX(H.N).
285
            AT, AB, DELF IN(M, M)
         CONTINUE
290
                 ...
                          END OF M-LGOP+**
                          CALCULATE ZONING AND INITIALIZE CELL CCCADINATES
          X : 13-0.
          J = 1
DG 390 P=1+ANTRES
NZOM=NZOMES(#)
            C
 345
110
                DX=2,=TH(H,R2)/FH-DE/FIH(H,R2)

Jh-J+CE(LSIF,R2) $ J1-J+1

CO 365 1-J1-Jh

x111-x11-11+CX
 395
                CH-RATICOCX+FI
 345
                     S JEAD (PIOJ S JOJE)
              aL+L
 340
          x(3)-X(3-1)
          ......
          IF (MATEL .ME. C) 60 10 500
                        BACIATION CEPOSITION
SA,AL, ASPEC. AZ, AJ, AACLE
          25 48
          write tip, sola i, a spec, a 2, a 3, angle
abble-gcstangle/17, 24977993
          tp image .mt. c) 60 TC 4TC IF image .te. 01 60 TO 42C MAG-1 9 TERF-1.
          ARBITRARY SPECTOUP INPUT
REAC FO. 01.02.FLNR
MRITREEO.FOT AL.EA.FLNX
```

SUBROUTINE GENRAT (c. inced)

```
OC 415 5H=1,AHNU
         REAC 11,41,42,781(NH),43,44,61(NH),45,46
MRITE(18, 11)41,42,781(NH),43,44,61(NH),45,46
         EI (A+)=EI(N+) *ANGLE*FLLX
415
         AR=1 1 GC TC 430
C
        BLACK BGDY INPUT

RPAC=REE 5 NFNU=105

IF (NS .NE. 1) GG TO 424

READ 60; A(1); I=1,8)

BRITE (18,60; A(1); I=1,8)
420
         READ 12,(TBL(AF),AM=1,1C9)
WRITE (18, 12)(TBL(AF),AF=1,1C5)
         AR=1
         REAC 92,AL,AZ,TEMP,A3,A4,ECAL
MRITE (18,92)A1,AZ,TEMP,A3,A4,ECAL
ECAL-ECAL+AAGLE
         CO 428 NF=1,1C9
E1(NH)=ECAL+.OLO(1,-.9+(NH/ECC))
428
                      COPPLIATION OF ABSORPTION COEFFICIENT - AC
430
         1-9386
         00 44C P=1.APTRLS
CO 433 AF=1.1C9
         ACCAST = ACGA(#)
433
         CO 445 NC+1.NCONST
         440
        1 )/ANGLE
445
C
C
         CONTINUE
                      DISTRIBUTE ENERGY INTO CELLS
          JEROP-JENO(#)-1
         DO 455 J-JREG.JBNDF
         ESLR=Q.
Cx=x(J+1)-x(J)
         00 43c RH=1,AMAU
IF (EI(AM) .LT. 1.E-20) GO TO 45c
EIZ=EI(AM)*(LL-EXP(-1.*AC(AM)*OT))
         ET (AH)=ET(AH)-ETZ
ESUM=£1Z+ESUM
 450
          CONTINUE
         SS(J,MS)=ESUMP4.18687/RMD(M)/DX/ESSTCPTNS)-STAMT(AS))+SS(J,AS)
JBEG-JBNO(M)+L
455
 440
         1+84-84
         IF (NR LE. MRAJ) GO TO 424
ENC OF RSPEC LOOP
CONTINUE 8 GO TO SCC
 445
                      ARBITRARY CEPOSITION
         REAC 50,14(1),1-1,71,E1(1)
WRITE (10.50)(4(1),1-1,71,E1(1)
REAC 70,41,42,14C(1),P-1,4P74(5)
BRITE (10.76)41,42,14C(1),P-1,4M74(5)
 470
         JEEC+1
DO 480 P-1, RPTRLS
JBECP-JERD(P)-1
         00 475 J-18EG.38809
CR-4(1413-4(1)
          E12-E111)+(1,-EXF(-AC(H)+OX))
          519-(1)13-(1)18
 475
          $$1J.11=E1E=4.-184E1/#HO(#)/D>/155TCP(#$)-$T##T(#$)1
 480
          1+(41048L-358L
                         MEAD IN PLCTYING CONTROLS -----
         00 502 1-1.4
 302
          IPPLCTIII-0
          PRINT 65
         # 1194(T+194(T+1) $80.163,164
1194(T+194(T+1)+1594(T+2)+1 4
1194(T+1) 8 1194(T+5+194(T+5)
 563
                                                          IFFICTIBLE FROTENIST IFFICTION
                                                          SPEAT SECREPLET
```

```
CO 504 1-1.9
           INISTEL INI
          INITITIE

GO TC 580

REAC 2G,A1.(IFFLCT([',!=1,6],A2.LINES
FRINT 2O,A1.1FPLCT([),!=1,6),A2.LINES

IF (IFPLCT([) .EC. C) GC TC 5LC

FEAC 21,A1.A2,ITPLCT.A3,NTPLCT.A4,IFLAGT

PRINT 21,A1.A2,ITPLCT.A3,NTPLCT.A4,IFLAGT

IF (IFFLCT(2) .EC. O) CO TO 580

REAC 22,A1.A2,(IFIST([],!=1,6),A3,(IHIST([]),!=7,9),A4,IFLAGJ

PPINT 22,A1.A2,(IHIST([]),!=1,6),A3,(IFIST([]),!=7,9),A4,IFLAGJ
509
510
Ç
         LPECK FCR
MEAC 6C,A1
IF (ECF,101 600,583
bRITE (18,55)
READ AA
                          CHECK FOR THE END OF DATA DECK
580
583
585
           IF (ECF. 10) 100.565
C
                   ### INITIALIZE CONCITION.CENSITY, TENSILE AND VIELD VARIABLES
400
          J=C
                                                                                                                              12/17
601
                                                                                                                              12/17
602
           TH- TEHE (M)
          11=J+1
EC 010 J=J1,JA
          CHE(1)=C(1)=C)
OHE(1)=C(1)=RFC(N)
          H4-111-PH
H4-12-L1H
           TELSIATERS (M.1)
           THE CALL TOR
           $ME(1)=0(1)+(X(1+1)-X(1))
           ASPPI-ASPERICE
           GO TO (410.41C.4CC.4CC.4CG.4CG.4CC.4CC) ASPMI

BEPIJPRIX(P,T) S NET(J)-RIX(P,E) S GC TC 61C

MERKIJPRIX(P,S) S NET(J)-O. S GC TC 610
606
607
610
           CONTINUE
          CONTINUE
HILL, 21-ALRTER
T(L) ALRES
HILL, ALLA ER
CONTINUE
           MIL.ZI=MIJFIA.ZJ-SFALL
           ETAM=1.6-12
3F (RATFL) #15,700,600
                          DEPOSITION EDIT
          700
              EPG-Q.
                 DC 702 ASALINSPER
              BU FOR ASALMSSATSSTOPINGS=~\APTINGS)+EPG

EPG=SSIJ,NSSATSSTOPINGS=~\APTINGS)+EPG

EPGTEGEGSTGIPS .GT. 1.ETS JSSAM-J

STORE ENERCY TERGS/GPS, CALCRIES AND SUM OF CALORIES IN ~A~
AIJ+SQQI~EPG & AIJ+ECCCI+EPC/ALREE)
702
£
£
          TIPESCONT S SEENSWOOMS OF THE STANDARD
103
           CONTINUE
```

```
P=K=J1=1
JZ=PING(JFIN-1.5C+K.JENC(P))
710
                     IF (J2 .NE. 50*K) GC TO 716
K=K+1 : WRIFE (18.44)(CISCPT(I).I=1.IC)
IF (J2 .NG. JANUEN)) GO TO 710
N=M+1 : MRIFE (18.69) : GC TO 710
718
                                                    INITIALIZE VELCCITY
£00
                     LISTANIOASL#SALSL
                    00 810 J=1.JFIN2
a10
                    JSTAR-JF(N203 A SDLRM=1. S GC TC 818
H(1-2)=PIRRCH S JSTAR=3 S SCLAP=1.
                     U(1)=.5+UZERG
........
                                                   VELCCITY ECIT
216
                     WRITE (18, 16) (515CPT(1),1=1,1C)
                    MIRL. INC 059 03 (LIX-(LIX-(LIX-
820
                     M=K=J1=1
                     J2 = PIRC( # 18-1-500#, JEAD(P))
825
                 MRTTE (18,17)(J,A(J),R(J),U(J),THL(J),C(;),C(J),C(J),T(J),ZHL(J),

1 MATE(L,M),MATE(Z,P),(H(J,I),I=1,J),J,J=J1,J;

IF (JZ .EC. JFIN-1) CO TO SCC
                   J1=J2+1
830
900
                     RETURN
                    FERNAT (A7, 13, A7, 13, A7, 13, A7, 23, E1C. 2, AE, 24, E1C. 3)
FORMAT(STAT, A8, E1C. 3))
10
11 12 14
                    (E.UI-3) TK-437
(V)IA35-1-1-154KREO
                 1 J (DX A(J) 261
                                                                                       U) EPC SEPCAL WHELL)
ZHERUR PATERRAL CORD
                                                                                                                                                                                                                          CLIE
                                                                                                                                                                                                           3
                                                                                                                               ERGS/GP CALS CYA/CP2 CM/S
                 3 /EC2F CH CH E4
4EC GA/CH3 DYA/CH2 CH/CH2
                    FORMAT 114, 9E1C.3.23.23.231C.313.811.15)
FORMATELINI.10-10//
14
                                                                                                                                                                                                                                             131H
                 1 3 OX 2 1(J)
                  PCTRRIGING, 10-107/

2 TGJ DX AGJ) UTJ) PHILD (GJ) CI
2 TGJ) ZMLGJ PATERIBL CCAC J
7 /102H CM CM CM/SEC DYA/CM2 CR/SEC
4M3 DYM/CM2 GP/CM2
FORMAT (29H TIME TG CCPPLETE GERRAT 15 F1C-3, 9H SECCHOS.)
FORMAT (A10-ALMI-12). AM-11)
                                                                                                                                                                                                                           C(j)
                  7 /1024
 10
                    FORMAT (Alg., elsk., 12). A9, 11)
FORMAT (Alg., elsk., 12). A9, 11)
FORMAT (Alg., elsk., 12);
FORMAT (Alg., elsk., elsk.,
31
31
31
35
                    FORMAT (11C)
FORMAT (11C)
FORMAT (11H)
FORMAT (11H CATE = 410)
FORMAT (2A4.12.2(A2.12).42.44.44.(22.42)
FORMAT (A1C.12.641C)
FORMAT (A1C.12.641C)
43
PCPPATITAS, ELQ. 31
                    FORMAT (AL, AC, TALQ)
FORMAT (BALG)
FCREAT (TALO. 285)
FORMATION, 12. A2. 15. 2(2A5. ELC. 3);
                     PORPAT (#8.#7.15.362#1.6(C.)11
```

```
70 FORPAT (1MO)
72 PORMAT (3A10,110,2(A10,E10.3))
73 FORNAT (2(A10,110),A10,E10.3,A10,110)
74 FORMAT (4(A5,110))
75 FORMAT (2A5,510.3)
77 FORPAT (2A5,5F10.4)
78 FORPAT(2A5,1415)
79 FORPAT(10X,7E10.3)
80 FORMAT(30X,1415)
91 FORMAT(3A10,F10.3,4(A8,12))
92 FORMAT(4(A5,2X,E10.3))
93 FORMAT(4(A8,2X,E10.3))
94 FORMAT(38HOERRCR ENCLUNTERED REACING INFLY DATA.)
END
```

SUBROUTINE GENRAT (concluded)

```
SUBRCLTINE HAFSTEF
        (INCLUDES ENTRY HSTRESS)
• CALLED BY HYDRE TO CEPPUTE X, L, C, E, R, S, P FCR THE
Ć
                HALFSTEP POINT BETHEER J AND J+1
        REAL PATE, MU. MUM. MUP. NEM. MET, MEC.
        INTEGER ", PGRCLS, FINTER, SCLIC, SFALL
                     MISCELLANEGUS
C
        COMMON CEF.CKS,DAVG,DELTIH,DGLD,DRHC,DTPIN,CTA,CTAH,DU,DX,EDLD,F,
FIRST,J,JCYCS,JINIT,JFIN,JSMAX,JSTAR,JTS,LSLB(11),MAXPR(11),A,
           NCYCS, NPERN, POLC, RLAST, SLAST, SMAX, IF, TIPE, TJ, TS, ULAST, UCLL,
           XLAST, XNCh, XCLD
                     EQUATION OF STATE - SCLID
        COMMON COSCIO,6),C1(6,6),C2(6),EPELT(6,5),EGSTA(6),EGST(.6),
FCSTC(6),ECSTE(6),EGST '6),EGSTH(6),EGSTN(6),EGSTS(5),MATL(2,6),
           PU(6),RHO(6),RHCS(6),TEAS(6,3),YACC(6),YO(6),JENC(6),NPOR(6,2),
           P. PUP, NHTRL S
                    EQUATION OF STATE - POFCUS
        COMMON AK(6), MUP(6), PORA(4,3), PEPS(4,2), PORC(4,3), RHGP(6,3),
       1 SACEP (4, 31, OREF, RECPV, PF, AC
C
                    RACIATION CEPGSITION
        COMMON SS(300,5).SSTOP(5),START(5),SDURP,SSTCFP,ASFEC.SSJ.JSS
                     CCORCINATE ARRAYS
        COMMON X(300),C(3C0),CHL(3C0),D(3CC),DHL(300),EHL(300),H(3CC,3),
NEM(300),NET(3C0),P(300),FHL(3CC),P(3CC),S(3C0),SFL(300),T(3CC),
       2 UE3001, UHE (3021, YHE (3001, ZHE (300)
        HALFSTEP VALUES
CORMON CH.DHLAST, CUP, EF, PP. RF. RFLAST, SH, SFLAST, UH, UH! AST, XH, XHLAST
C
        IDENTIFIERS
COMPGN DISCPY(101, ICENT, JEDITIZC), JREZON(151, NOATE(3), MEDIT, NEDTP.
C
       1 NJEDIT. NR. RREZON, NSEPRAT, NSPALL, ATEDT, ATEX, NTRESSI, TECITISOI
        CCHOITION INLICATORS
COMMON INF,LINJER,MIRROR,NCRMAE, FGRELS, MINIER, SULIO, SPALL
¢
        SPALL AND RELAXATION
COMMON NSR16), TSR16,6), USP150), ESP(50), LPLX, NICK, NEPH, NETH
Ç
C.
        CIPENSICH LILO)
        CO 1 1=1,10
L(1)=0
         CC=0.0
        Dxax(J+1)-x(J) & EDLD=EHL(J) & UHH=.5+(U(J)+U(J+1))
CCLD=BFL(J) & G2=0.
         # (C(J+1)*C(J)*U(J+1)*U(J)*C2(P) .EC. C.) 7,4
        62= C2(M3*ABS(C(J+1)-C(J))*RHC(M)/(C(J+1)*C(J)*DCLD)
1F ((1.5*DTM-D1NM)*(1.5*DTNH-DTA)) 15.1C.1G
10
         LH=UHL (J)-.5+((DTAH+CTA)/DX+G2)+(R(J+1)-R(J))/COLC
        GC TE 18
        EMPERME .50(CINH/DX+67)0(R(J+1)-R(J))/ECLC
15
         CHL(J)=DH=ZHL(J)/(Lx+.5*DTNF*(U{J+1)-U(J))]-DCLD*G2*(U{J+1}-U{J)}
         HJ=CLEJHJ
        XH=.5*(X1.1+1)+X(J1)+.2**D*N+*(L+H+LH)
        EPALLIMENTATION -- POPERTY NOT STATE -- COLOR-1-LIE EPALLIMENT STATES CONFLICTION
        ENTRY HSTRESS
                                                                                                     ENTRY
        FUFMELT(EN.H)
         IF 10H1 20.20.25
20
         PRIAT 110.A.J.DF.TIME
        PRINT 110-1-3-5-7-1 IRE
CALL ECDUMP & LSUB!73-1 & N=N-1 & CALL SCRIBE & STOP
GORPUTE CLC VALUE OF SC

SOM=SPL[J]-PNL[J] & TF=1.-EH+ECSTAIP) & CREF-CH+TF
ESTABLISH ROUTE

IF IHIJ:10 .NE. PORCUS) GO TO 5C

IF IF .GT. 0.) GC TC 4G
UNCONSQLIDATED
        MUR-C. S CALL ECST (EN-DH, PHLIJ). P. CC1
         SOMMRE-B.
        3DH=RC=0.

IF :PHL(J) .LT. Q.) 37,39

PHL(J)=SML(J)=C. 8 CO TC 9C

H(J,1)=SOLID 8 L(4)=39 8 C

SELECT POROLS REGION
39
                                                GE TC 53
40
        ANDPV-F+(RHCP(5 .PP)-RKCS(F))+FHCS(H)/TF
         L131-40
```

SUBROUTINE HAFSTEP

```
IF (DH .GT.RHOPY) 49,43
43
          NC=0
          L(41=43
45
           NC = NC+1
           IF (DREF .GT. RHOP(NC+1,MP) .AND. NC .LT. NPOR(M,2)) 45,60
                         SOLIO
          H(J+1) = SOL 10
49
           MUM=MU (M) +F
50
          CALL EQST(EH,DH,PHL(J),M,CHL(J))
L(5)=50
           YADDM=YADD(M) $ NC=5
IF (NPOR(M+1)) 55.55,54
DREF=AMAX1(DREF,RHDP(5,MP))
           IF (DREF-RHOP(6, MP))67,55,55
          COT=COSQ(M.6) $ CIT=CI(M.6)
55
           L(6)=55
          L(6)=>>
GO TO 70
POROUS
60
          CALL POREQST (EH, DH, PHL(J), M, MP, CHL(J))
           YADDM=YADDP(NC.MP)
          COT=COSQ(M,MC)+(COSQ(M,MC+1)-COSQ(M,MC))+(DREF-RHOP(NC,MP))
L /(RHOP(NC+1,MP)-RHOP(NC,MP))
CIT=C1(M,NC)+(C1(M,NC-1)-C1(M,NC))+(DREF-RHOP(NC,MP))
67
             /(RHOP(NC+1,MP)-RHOP(NC,MP))
          L(6)=67
                         VELOCITY GRADIENT AND TIME STEP PARAMETERS
          DUM=-2.*(DH-DDLD)*(X(J*1)-X(J))/((CH+DDLD)*DTNH)
DUMM=AMIN1(DUM,0.) $ CS=CHL(J)-DUM/2.
CF=C1T-COT*DUMM/CS $ CEF=CS*(1.+CF*(1.+.5*CF))
70
          CF=C1T-C0T*POUMM/CS $ CEF=CS*(1.+CF*(1.+.5*CF))
DEVIATOR STRESS

IF (NSR(N) .EQ. 0) GO TO 78
DRHD=OH-DOLD $ DAVG= (DH+DOLD)/2.
SDO*SOH $ COEF=MUM*DRHO/DAVG $ MEDW=2.*MUM
NSRM*MSR(N) $ CNUM=MUM
GO TO (73,74,74,73,75) NSRM
CALL RELAXISOH,H(J,3),YHL(J),SDO*DTNH,TSR(M,1),ORHO*COEF,YADDM,N.
L J*(SR(M,2),YO(M),NSRM)
C
7.4
           GD TO 80
74
          CONTINUE
         CALL BANDRLX(SDH.H(J.3), YRL1J), SDO, DTNH, DRHO, COEF, MEUH, YADDM, N.J.
1 TSR(M,1): TSR(M,2), TSR(M,3), TSR(M,4), TSR(M,5), TSR(M,6), NEM(J),
2 NET(J), NSRM)
           GO TO 80
          CALL BAUSCHI(H(J,3),SDD,SDH,YHL(J),YO(M),YADDM,GRHO,COEF,TSR(M,1),
TSR(M,2),MUM,N,J,DAVG,TSR(M,3),TSR(M,4))
          CJ=CHE(J) S CSQ=CJ=CJ=AMAX1(O.,\MUM-DMUM)/DAVG:
CQ=CSQ=UJ=CJ S CHL(J)=CSQ+CJ/CQ+.25=CQ/CJ
           GO TO BO
78
           SOH=504+2. *#UM+(OH-DOLD)/(DH+DOLD)
           1(7)=78
           IF (ABSISDH) .LT. YHL(J) MF) GO TO BO
YHL(J) MANINI(ABSISDH), YHL(J) MADDHMANS(DH-DOLD))
           SOH-SIGNIVIL (J) F. SOH)
           L(81=79
          99
100
          ##?TE (18-5000) J.M.EH.DH.PHL(J).SDH.(L(I).1-1-8)
##ITE (18-500) CHL(J).CS.CEF.COT.C1T.RH
           RETURN
120
          PORMAT (20H STOP IN HAFSTEP, Nº14,4H, J=14,4H, D=10,3,

TH, TIME=E10,3;

FORMAT (15H HAFSTEP, J=13,4H, Nº14,4H EH=E10,3,4H DH#E10,3,

4H PH#E10,3,5H SDM#E10,3,3H L=013;

FORMAY (6H C=E10,3,5H, CS#E10,3,6H, CEF#E10,3,6H, COT#E10,3,
9000
500:
              6H. CIT-E10.3,5H, RH-E10.3)
           END
```

SUBROUTINE HAFSTEP (concluded)

```
SUBROLTINE HYDRO
               SUBROUTINE CONTROLS THE MAIN CALCULATICA CYCLE
                                CONTAINS 4 PATHS
                                         1. NORMAL - COORDINATES BITHIA PATERIAL
                            2. INTERFACE — INTERFACE ESTAGES MATERIALS
3. INTERFACE SPALL — SEPARATED INTERFACE SETWEEN MATERIALS
4. SPALL — COURDINATE AT WHICH SEPARATION HAS OCCURRED CALLS HAFSTEP FOR HALFSTEP CALCULATIONS AT EACH COCCREMATE CALCULATES X, U, D, E, AND CALLS JSTRESS TO DETAIN STRESSES CHECKS FOR SPALLING AND RECOMBINATION
                                CEMPLIES PINIPUP PEAPITTEE TIME STEP FOR NEXT CYCLE
                        • SET DOLD, EOLD, DU, P(J), S(J) IN FREPARATION FOR JSTRESS.
               REAL MATL, MU, MUM, MUM, NEM, NET
INTEGER M, POROLS, RINTER, SGLID, SPALL
                INTEGER HJ.HJ3
                                        MISCELLANEOUS
               COMMON CEF.CKS.DAVG.DELTIM.DCLC.CRPC.CTPIN.CTN.CTNH.DU.CX.EGLD.F.
FIRST.J.JCYCS.JINIT.JFIN.JSMAX.JSTAR.JTS.LSL8(11).MAXPR(11).N,
                   NCYCS, NPERN, PCLD, RLAST, SLAST, SPAX, TF, TIME, TJ, TS, ULAST, UOLC.
             3 XLAST . XACH . XCLC
               EQUATION OF STATE - SCLID

COMMEN COSCIO,6),C16,6),C2(6),EMELT(6,5),EGSTA(6),EQSTC(6),

EQSTD(6),EQSTE(6),EQSTG(6),ECSTH(6),EGSTA(6),EGSTS(6),MATL(2,6),
                  RL(6),RHO(6),RHCS(6),TENS(6,3),YACC(6),YO(6),JBND(6),NPCR(6,2),
                     M. NUM, NMTRLS
                                         EQUATION OF STATE - FORCUS
                COMMCN AK(6), MUP(6), PORA(4,3), PORB(4,3), PGRC(4,3), RHCP(6,3),
            1 YAECP14,33,DREF.RPCFV.PF.RC
RACIAT3CN CEPOSITION
COMMON SS(300,5),SSTOP(5),STAR1(5),SDURP,SSTCPP,RSFEC,SSJ,JSS
CCCRDINATE ARRAYS
               COMMON X(300),C(360),CHL(3GC),D(3CC),QHL(300),EHL(300),H(3CC,3),
                     NEM(300) . NET(300) . PE300) . FHL(300) . BE300) . SE300) . SE300) . SE300) . TE300) . TE300)
             2 U(300), UHL(300), VHL(300), ZHL(300)
HALFSTEP VALGES
CONMON CH, DHLAST, CUF, EF, PF, RH, RFLAST, SH, SHLAST, UH, UHLAST, XH, XHLAST
                                         IGENTIFIERS
                COMMON DISCPT(10), ICENT, JEDIT(20), JREZCA(15), ADATE(3), NEDIT, NEDTM,
             1 NJEDIT.NR.NREZGN.NSEPRAT.ASPALL.ATEDT.ATEX.ATR(15).TECIT(50)
                                         CONCITION INCICATORS
               COMMON INFILINTER, MIRROR, MORMAL, FORCES, RINTER, SCLID, SPALL AND RELAXATION
               COMMON NSR161, TSR16, 61, USP(50), >SF(5C), LRLX, NICK, AEPH, NETH
                CIPENSIEN LILES
                CALL SECONDERNORD & DUR-PROM-FIRST & D7-DTFIR=1.
                 SMAX=0.
                                         QUIER HYDRC LOOP
                DC 1000 AN=1.NCYCS
                 TIME=TIME+OTHE
                JSPALL =0
                 M=MP=O
                CO 906 JMJ: "TTJJFIR
IF (LSUB(L) .GE. PAXFR(L)) CC TC 3
MRJTE (LB.5000) N.J.>(J.2)
ì0
                 CO 2 1-1,10
                 _(1)=0
                                         CHECK FOR THE APPROPRIATE PATH
                THE R FUN THE APPROPRIATE P.

BOLD=X(J) & UCLG=U(J) & RCLC=P(J)

IF (r(J,2) .EC. NORMAL) GO TO 1CC

IF (r(J,2) .EC. L(AYER) GC TC 9CO

IF (MIJ,2) .EC. RINTER) GO TO 2CC

IF (r(J,2) .EQ. SPALL ) GO TO 3CO

IF (H(J,2) .EC. MIRROR) 500,40C
```

SUBROUTINE HYDRO

```
HORMAL PATH WITHIN A MATERIAL
                                                                                                             ********
         IF (NSPEC .GT. 0 .DR. ABS(U(J)-U(J+1)) .GT. 1.E-3 .DR.
L ABS(RHLAST) .GT. 1.) GD TO 102
UM=U(J) 8 RH=R(J) 8 XH=.9*(X(J+1)+X(J)+DTHH+U(J))
100
        UM=U(J) $
L(1)=10%
101
          DH=DHL(J)
                         S EIPEHL(J)
         THE H-U(J)
         X(J)=X(J)+DTNH=U(J) & SSJ=SSCAL(J) & GO TO 800
         CALL HAFSTEP
102
          L(1)=102
         DOLD=.5*(DH+DHLASY) $ EDLD=.5*(EH+EHLASY)

$\frac{1}{2}.5*(PHL[]]+PHL[]-1)} $ \tag{YJ=.5*(YHL[])+YHL[]-1}}

$\frac{1}{2}.5*(SHL[])+SHL[]-1}
                       VELOCITY CALCULATION
C
         U(J)=UOLD-2.*OTNM+(RH-RHLAST)/(ZHL(J-1)+ZHL(J))
COORDINATE CALCULATION
Ċ
         X(J)=X(J)+.5+0YNH+(U(J)+UOLD)
         DENSITY CALCULATION
DIJ)=DGLD/(1.+.5=DTNH=(U(J+1)-ULAST)/(X(J+1)-XLAST))
C
                       ENERGY CALCULATION
         SSJ1=SSCAL(J)
103
          EJ=EOLD-.5*(RH+RHLAST)*(1./D(J)-1./OOLD)+SSJ+SSJ1
          551=5511
         STRESS CALCULATIONS

DU=UH-UHLAST 8 HJ=SON.ID

IF ((H(J-1,1) .Eq. POROUS) .OR. (H(J,1) .Eq. POROUS)) HJ=POROUS
C
          HJ3=2
         TI (NSR(M) .EQ. 0) GD TO 106

IF (NSR(M) .EQ. 0) GD TO 106

IF (ABS(S(J)-P(J)) .GT.YJ) MJ3=2-IFIX(SIGN(1.,S(J)-P(J)))

NSRM=NSR(M) 8 GO TO (106,104,104,106,105) NSRM

NEMH=.5+(NEMIJ)+NEM(J-1)) 8 NETH=.5+(NET(J)+NET(J-1)) 8GO TO 106

IL=MINO(H(J,3),H(J-1,3)) 8 TH=MX0(H(J,3),H(J-1,3))
105
             (((IH-IL-1)*(IH-IL-5).EQ.0).AND.(HJ3.EQ.2)) IL=IH
          HJ3=IL
106
         CALL JSTRESSIN.MP.J.EJ.HJ.HJ3.YJ)
          UHL [J-1]=.025#[U[J]+UNEW+UOLD+ULAST]+.9#UHLAST
          UWEN=U(J)
         C
                       CHECK STRESS AND SET INDICATORS FOR SPALL
         IF (R(J) .GT. TJ) GO TO 800
IF ((NSEPRAT .GE. 0) .DR. (F .GT. 0.)) GO TO 110
R(J)*S(J)*P(J)=0. $ GO TO 800
 108
110
          JFINN1=JFIN-1
          1 (2)=110
          IF (NSPALL .EQ. 0) 114,111
         10 113 114 304 07 1171...

10 112 JJ=1.JFINM1

1F (HIJJ-2) .EQ. 1) 113,112

CONTINUE $ NSP=H(J,2)=1 $ GO TO 115
111
112
          CONTINUE
          NSP=NSPALL=H(J,2)=NSPALL+1
XSP(NSP)=X(J) $ USP(NSP)=U(J)
IF (LSUB(9) .GE. MAXPR(9)) GO TO 120
 115
          DESC=10H SPALLED

WRITE (18.5115) DESC:N:NN:J:NSPALL:NSP:R(J):TJ:TIME

WRITE (18.5050) DT:R(J):S(J):C(J):D(J):T(J):U(J):UH:DH:EH:PHL(J):
            RH.XH.(L(1),1=1.5)
          WRITE (18,5051) DIMIN, DINH, U(J+1), X(J+1), X(J), CEF
         JSPALL =J
R(J)=S(J)=T(J)=0.
          GO TO 800
                        INTERFACE
          LEFT VALUES ARE IN (J-1) CELLS AND RIGHT VALUES ARE IN (J) CELLS
```

SUBROUTINE HYDRO (continued)

```
200
          MLAST=M & MPLAST=MP & M=M+1 & MP= NPOR(M+2)
          L(1)=200
          CALL HAFSTEP
          SSJ=SSCAL(J)
                         FIRST ESTIMATE OF UM, HALF STEP VELOCITY AT INTERFACE
205
                             J2=J-2
          UM=.5+(U(J)+U(J))-DTNH+(R(J+))-RLAST+4.+(RH-RHLAST))/
             (ZHL (J)+ZHL (J2))/6.
          SSJ1=SSCALIJI)
          IF (LSUB(10) -GE, MAXPR(10)) GO TO 209
LSUB(10)=LSUB(10)+1
WRITE (18,5002) N,MLAST,M,UM,U(J),UM,UHLAST,RH,RHLASY,RLAST,R(J)),
              R(J).R(J+1)
          ITER=0
209
          ITER=ITER+1
210
          S(J)=SHL(J) $ P(J)=PHL(J) $ S(J1)=SHL(J2) $ P(J1)=PHL(J2)
                          RIGHT SIDE ENERGY AND STRESS CALCULATIONS
          DU=2.*(UH-UM)
DOLD=DH $ EOLD=EH
D(J)=DH/(1.+DTNH: (UH-UM)/(X(J+1)-X(J)))
          EJ=EOLD+RH*(1./UJLD-1./D(J))+2.*SSJ
NEMH=NEM(J) $ NETH=NET(J)
          CALL JSTRESS(M, MP, J, EJ, H(J, 1), H(J, 3), YHL(J))
C
                         LEFT SIDE ENERGY AND STRESS CALCULATIONS
          DU=2.+(UM-UHLAST)
          DOLO=DHLAST $ EOLO=EHLAST
D(J-1)=DHLAST/(1.+DTNH=(UM-UHLAST)/(X(J-1)-XLAST))
          TJ=EDLD+RHLAST*(1./DOLD-1./D(J-1))+2.*SSJ1
NEMH=NEM(J2; $ NETH=NET(J2)
CALL JSTRESS(MLAST,MPLAST,JR,EZ,H(J2,1),H(J2,3),YHL(JZ))
IF (LSUB(10) _GE. MAXPR(10)) GO TO 211
          WRITE (18,5003) [TER,UM,UM1,UM2,UM3,R(J),R(J1),RJ11,RJ11,RJ2,RJ12,
              $(1),$(11)
          CONTINUE
          CONTINUE

[TERATION TO COMPUTE INTERFACE VELOCITY AND STRESS
IF (ITER .G8. 10) 224,213
RS=ABS(R(J))+ABS(R(J!))
IF (RS .LT. 1.E6) 225,215
IF (ABS(R(J)-R)1)1/RS .GT. .01) 216,225
IF (ITER-2) 220,219,217
IF (ABS(R)2-R.J.E) .LT. ABS(R)1-R)11)) 218,219

MULTIPLE ITERATION ROUTE

UM3=UM2+(UM-UM2)+(R)12-R)21/(R(J)-R)2-R(J1)+R)12}
UM3=UM2+(UM-UM2)+(R)12-R)21/(R(J)-R)2-R(J1)+R)12}
C
213
216
217
218
          UCON=AMAX1 (.20(ABS(UH)+ABS(UHLAST)+ABS(UM1))*FLOAT(ITER*#2).100.)
          UM=UM3=AMAXI(UM1-UCON,AMINI(UM3;UM1+UCON))

RJ1=R(J) $ RJ1=R(J1) $ GO TO 210

2NO AND MULTIPLE /TERATION ROUTE

UM3=UM1+(UM-UM1)+(RJ11-RJ1)/(R(J)-RJ1-R(J1)+RJ11)
219
           MU= SMU
          UCON-AMAXE(.2+(ABS(UH)+ABS(UHLAST)+ABS(UH2))+FEDAT(ITER#+2),100.)
          UM=UM3=AMAXI(UM2-UCON, AMINI(UM3, UM2+UCON))
          RJ2=R(J) $ RJ12=R(J1) & GO TO 210
FIRST ITERATION ROUTE
220
          IF {(UM-UH)*(UHLAST-UH) .EQ. 0.) 225,222
UM*UM-(R{J})-R{J}11}/(D{J}+C{J}+D{J}1+C{J}1)
222
          UM=MAXI(AMINI(UM,1.=10+AMAXI(UM,UMLAST)),0.90*AMINI(UM,UMLAST))

MILL=R(JI) $ RJI=R(J) $ UMZ=UM $ GO TO 210

MRITE (18,5224) N.MLAST,M

FORMAT (44H 'NTERFACE ITERATIONS DID NUT CONVERGE AT N=14,

1 12H BETHEEN M=12,7H AND N=12)

MRITE (18,5003) ITER,UM,UM1,UM2,UM3,R(J),R(J1),RJ1,RJ11,RJ2,RJ12,
224
5224
226
         1
              $(1),$(1))
                          END OF INTERFACE ITERATION
¢
          COGRDINATE CALCULATION
U(J)=U(JL)=U0LD-2.+DTNH+(RH-RHLAST)/(ZHL(J2)+ZH((J1))
225
```

SUBROUTINE HYDRO (continued)

```
C
        UHL(J2)=.025*(U(J1)+UNEW+UOLD+ULAST) + .9*UHLAST
C
        CHECK STRESS AND SET INDICATORS FOR SPALL IF (R(J)) .LT. T(J1)) 230.800
        H(J,2)=SPALL
L(2)=230
230
        IF (LSUB!9) .GE. MAXPR(9)) GO TO 235
DESC=10H INT SPALL
         WRITE (18,5230) DESC,N,NN,MLAST,M,R(J1),T(J1),TIME
         CALL EDDUMP
235
        R(J)=R(J1)=S(J)=S(J1)=T(J1)=0.
        GD TO 800
C******
                     INTERFACE SPALL
                                                                                                   ********
        IF (J .EQ. JINIT) 330,310
MLAST=M $ MPLAST=MP
L(1)=310
300
310
                    LEFT SIDE
         J1=J-1 $ $$J1=$SCAL(J1)
%LOLD=X(J-1) $ ULOLD=U(J-1)
         U(J-1)=ULOLD+2.*DTNH*RHLAST/ZHL(J-2)
         D(J-1)=DHLAST/(1.*DTNH*(.5*(U(J-1)+ULOLD)-UHLAST)/(X(J-1)-XLAST))
X(J-1)=XLOLD+.5*DTNH*(U(J-1)+ULOLD)
         UHL(J-2)=.25*(U(J-1)+UNEN+ULOLD+ULAST)
IF (J .EQ. JFIN) 900,330
RIGHT SIDE
 330 M=M+1
MP=NPOR(M,2)
         L(2)=330
         LTC:=350

IF (ABS(U(J)=U(J+1)) .LT. 1.E-3 .AND. NSPEC .EQ. 0) 331,332

UH=U(J) $ RH=R(J) $ XH=.5*(X(J+1)+X(J)+DTNH*U(J))

X(J)=X(J)+DTNH*U(J) $ DT=1, $ SSJ=SSCAL(J)
331
         DH=D1_[J]
         IF (J .EQ. JINIT) 800,335
         CALL HAFSTEP
         SSJ=SSGAL(J) $ UDLD=U(J) $ XOLD=X(J)
U(J)=UDLD-2.*DTNH*RH/ZHL(J)
         D(J)=DH/(1.+DTNH+(UH-.5+(U(J)+UOLD))/(X(J+1)-X(J)))
         X(J)=XOLD+.5*OTNH*(U(J)*UOLD)
OT=(X(J+1)+U(J+1)*DYNH-C(J))/CEF
         IF (DT .LT. 0.) OT=1.
         UNEW-U(J)
                     CHECK FOR RECOMBINATION
C .
         IF (J.EG. JINIT) GO TO 800
IF (XIJ) .LE. X(J-1) 361,800
IF (LSUB(9) .GE. MAXPR(9)) GO TO 365
DESC-10MINT RECOMB
361
         MRITE (18,5230) DESC.N.NN.MLAST.M.R(J-1).T(J-1).TIME
RESET ARRAY VARIABLES AND GO TO INTERFACE ROUTE
M[J.2)=RINTER & X(J)=XOLD & X(J-1)=XLOLD & U(J)=UOLD
         U(J-1) = ULOLD $ GO TO 205
                     SPALL WITHIN A MATERIAL
          LEFT VALUES ARE IN J-CELLS AND RIGHT VALUES ARE IN XSP AND USP
 400
         MSP+M(J,2) $ XOLD-XSP(MSP) $ UOLD-USP(MSP)
                             T(J)=0. $ R(J)=0. $ DX=X(1+1)-XOLD
         INCOLD SULL)
```

SUBROUTINE HYDRO (continued)

L111-400

```
C
C
                     PREPARATION FOR HSTRESS IN SPALL
C
         CALL HSTRESS
         USP(NSP)=UDLD-2.*NTNHMRH/ZHL(J)
USP(NSP)=XDLD+.5*DTNHM(USP(NSP)+UDLD)
D(J)=.5*(DH+DHL 3T)
DT=(X(J+1)+UJ+1)*DTNH-XSP(NSP))/CEF
         IF (DT .LT. 0.) DT=1.
UHL(J-L)=.25*(ULJ:)*UNEH+UOLD+ULAST)
         UNEW=USP(NSP)
                      CHECK FOR RECOMBINATION AND RESET QUANTITIES AT J
C
         IF (XSP(NSP) .LE.X(J)) 410,800
         H(J,2)=NORMAL
L(2)=410
410
         X(J)=-5+(X(J)+XSP(NSP))
U(J)+-5+(U(J)+USP(NSP))
         D(J)=.5+(ZHL{J-1)/(X(J)-X(J-1))+ZHL(J)/(X(J+1)+DTNH+U(J+1)-X(J)))
         IF (NSP .EQ. NSPALL) NSPALL=NSPALL-1
IF (LSUB(9) .GE. MAXPR(9)) GO TO 420
         DESC=10HRECOMBINED
         DESC=10HRECOMBINED

WRITE (18,5115) DESC,N,NN,J,NSPALL,NSP,R(J),TJ,TIME

WRITE (18,5050) DT,R(J),S(J),C(J),D(J),T(J),U(J),UM,DM,EM,PHL(J),

RM,XC(T([1),I=1,5)

WRITE (18,5051) DTMIN,DTNH,U(J+1),X(J+1),X(J),CEF

RETURN TO NORMAL PATH FOR STRESS CALCULATIONS

P(J)=S(J)=0. $ DOLD=.5+(DH+DHLAST) $ EOLD=.5+(EH+EHLAST)
420
          YJ-.5+[YHL(J]+YHL(J-1))
         IF 10.3*DT .GT. DTMIN) GO TO 103
DTMIN=0.3*DT & JYS=J & GO TO 103
                      MIRROR AT FRONT SURFACE
C......
                                                                                                            ******
 500
         M-M+1 $
L(1)=500
                   $ MP-NPOR(M.2)
         CALL HAFSTEP
         DGLD=DH $ EOLD=EH $ DU=2.*(UM-U(J))
D(J)=DOLD/(1.*DTNH+(UH-U(J))/(X(J+1)-X(J)))
          SSJ=SSCAL (J)
         EJ=EOLD+RH+(1./DOLD-1./D(J))+2.455J
         S(J)=SHL(J) & P(J)=PHL(J)
NEMH=NET(J) & NETH=NET(J)
         CALL JSTRESS(M.MP, J. EJ. H(J. 1), H(J. 3), YHL (J))
          R[J]=2.*R[J]-S[J]
         (L)U*HMTQ+(L)X=(L)X
          DT=(X(J+1)+DTMH+U(J+1)-X(J))/CEF
          UNEN-U(J)
          IF IRIJI .GT. TIJFIN-11) on to 800
          HIJ.21-SPALL
         1121-530
         DESC - IOH MIR SPALL
         WRITE (18.9230) DESC.N.NH.M.N.R(J).T(JFIN-1).TIME
P(J)-R(J)-S(J)-T(JFIN-1)-0.
 *******
                      END OF MIRROR PATH
         CONTINUE
         L{4}-800
                      END OF CYCLE RESET
         MLAST=XOLD & ULAST=UOLD & RLAST=ROLD
MRLAST=XH & UHLAST=UH & RHLAST=RH &
EHLAST=EH 3 SHLAST=SH
                                                   RHLAST-RH & DHEAST-DH
```

SUBROUTINE HYDRO (continued)

```
SMAX CALCULATION
IF (S(J) .GT. SMAX) 820.822
SMAX=S(J) S JSMAX=J
420
         TIME STEP CALCULATION IF (DT .LT. DTMIN) 824.826
822
         DIMIN-DT & JTS-J
824
         DTP=DT
         IF (LSUB(1) .GE. MAXPR(1)) GO TO 850
WRITE (18.5050) DT.R(J).S(J).C(J).D(J).T(J).U(J).UH.DH.EH.PHL(J).
       1 RH,XH,(L(1),1=1.5)
MRITE (18,5051) DTMIN,OTNH,U(J+1),X(J+1),X(J),CEF
        JSTAR CALCULATION

IF (ABS(U(J)) .LT. 1.E-3) 851,900

IF (J .GT. JSTAR) 852,900

JSTAR=J-I
GO TO 000
850
852
         GO TO 990
                       END OF HYDRO INNER LOOP
900
         CONTINUE
         JSTAR=JFIN-1
990
         DTN-OTHH
IF (JSPALL+NSEPRAT .LE. 0) GO TO 1000
JINIT-JSPALL $ H(JINIT,2)=SPALL
C END OF YYDRO OUTER LOOP
1000 CONTINUE
          IF (LSUB(8) .LT. MAYPR(8)) 1001,1002
         CALL SECOND(THIX) S DUR-THIX-XMON
MRITE (18,5010) N.DUR
1001
         LSUB(81=LSUB(8)+1
        RETURN
FORMAT( LOHOHYDRO SEGIN. N=14,3H J=14,8H H(J,2)=RL)
FORMAT (21H HYDRO, INTERFACE, N=15,4H, H=12,5K AND 12,5H, UM=
1 E10.3/ 9H U(J)=E10.3,4H UH=2E10.3,4H RM=2E10.3,
/128H
1005
5000
5002
       2 4M, R+4E10.3
3 ITER UM
4 RJ1 RJ11
                                           UN2 UM3 R(J) :
RJ2 RJ12 S(J) S(J-1)
                                      UMI
         M SECONDS.//)
5115 FORMAY (IN ,A10,7H N, NN=214,4H, J=14,6H, NSP=213,7H, R, T=2E10.3, 1 7H, TIME=E10.3)
5230 FORMAT (IN ,A10,7H N, NM=214,4H, H=213,7H, R, T=2E10.3,7H, TIME=,
            £10.31
        FORMAT (10H HYDRO DT-E10.3,3H R-E10.3,3H S-E10.3,3H C-E10.3,

1 3H D-E10.3,3H T-E10.3,3H U-E10.3/ 7X,4H UH-E10.3,4H DH-E10.3,

2 4H EH-E10.3,4H PH-E10.3,4H RH-E1G.3,4H XH-E10.3,4H, L-514;
         PORMAT 110H
                                   DT=2610.3,44, UIJ+11-610.3,44, X+5-2610.3.
        t M. CEF-ELO.31
         FHO
```

SUBROUTINE HYDRO (concluded)

```
SUBROUTINE JSTRESS(MJ. MPJ. JJ. EJ. HJ. HJ3. YJ)
              CALL ED BY HYDRO TO COMPUTE R. S. P. AND TO SET T
                          CALLS EGST, POREGST, AND FRELT
TERMINATES THE PT GRAM FOR A NEGATIVE DENSITY
              REAL MATE, MU, MUM, MUP, No
                                                                             ET, ME OM
              INTEGER H, POROUS, RINTER, SOLID, SPALL INTEGER HJ. HJ3
                                    MISCELLANEOUS
              COMMON CEF, CK5. DAYG, DELTIM, DOLD, DRHO, DTMIN, DTN, DTMH, DU, DX, EOLD, F,
            I FIRST.J.JCYCS.JINIT.JFIN.JSMAX.JSTAR.JTS.LSUBIIII.MAXPR(III).N.
2 NCYCS.NPERN.POLD.RLAST.SLAST.SMAX.TF.TIME.TJ.TS.ULAST.UOLD.
             3 XLAST, XNOW, XOLD
            EQUATION OF STATE - SOLID

COMMON COSQ(6,6),C1(6,6),C2(6),EMELT(6,5),EQSTA(6),EQSTC(6),

1 EQSTD(6),EQSTE(6),EQSTG(6),EQSTM(6),EQSTM(6),EQSTS(6),MATL(2,6),
C
                 MU(6),RHO(6),RHOS(6),TENS(6,3),YADD(6),YO(6),JBND(6),MPOR(6,2),
            3 M, MUH, NATRES
                                     EQUATION OF STATE - POROUS
τ
              COMMON AK(6), MUP(6), PORA(4,3), PORB(4,3), PORC(4,3), RHOP(6,3),
            1 YADDP(4,31,OREF,RHOPV,MP,NC
              RADIATION DEPOSITION
CONNON_SS(300,5),SSTOP(5),START(5),SDURM,SSTOPM.NSPEC,SSJ,JSS
C
                                     COORDINATE ARRAYS
              COMMON X{300},C(300).CHL(300).D(300).DHL(300).ENL(300).HL(300).HL(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T(300).T
             2 U(300), UHL(300), YHL(300), ZHL(300)
              HALFSTEP VALUES
CONMON DH, DHLAST, DUH, EH, PH, RH, RHLAST, SH, SHLAST, UH, UMLAST, XH, XHLAST
IDENTIFIERS
C
            COMMON DISCPT(10), [DENT, JEDIT(20), JREZON(15), NDATE(3), NEDIT, NEDIT, NEDIT, NJEDIT, NR. NREZON, NSEPRAT, NSPALL, NTEOT, NTEX, NTR(15), TEDIT(50)
                                     CONDITION INDICATORS
C
              COMMON INFILINTER, MIRROR, NORMAL, POROUS, RENTER, SOLID, SPALL
                                    SPALL AND RELAXATION
              COMMON NSR(6),TSR(6,6),USP(50),XSP(50),LRLX,NECK.MERM,NETH
               DIMENSION L(10)
               DG 1 1-1-10
               1(1)+0
               CC .U.O
               IF (D(J31) 20,20,25
               PRINT 110.N.JJ.D(JJ).TIME
20
              CALL EDDURP & LSUBITY+1 & N=N-1 & CALL SCRIBE & STOP SDJ-SIJJ-PIJJ
23
               F-EMELTICATION & TF-1. + EURIATZOB-LB+. 1-TT & LLM. LB TJB-TF
              ESTABLISH ROUTE

IF IMJ .EQ. SOLID: 50.30

IF IF .GT. 0.3 GO TO 40
C
 30
                                    UNCOMSOLIDATED REGION
                                S CALL EQSTIFICATION PERSON MISCES
               知}件=()。
               L121-32
              L(2)*32

1F (P(JJ) .LT. 0.) 37.51

P(JJ)*S(JJ)*R(JJ)*0. 5 GO TO 85

SELECT POROUS REGION

8HOPY*F*(RHOP(5 ,MPJ)*RHOS(RJ) ) + RHOS(RJ)/TF
 40
               L111=40
                 (F (D(JJ) .GT. RNOPY) 50,43
     43
               NE -0
               MC+MC+1
 45
               IF (DREF ,GT, RHOP(NC+1, MPJ) .AND. NC .LT. NPORTHJ. (3): 45.40
PRESSURE, SOUND SPEED, TENSILE STRENGTH IN SOLID
               MIN-MU(MI) OF & CALL EQSTIBILITIES, PLIII, MI.CELII)
 50
               TADOM-YADO(RJ) 5 MC=5 8 TJ=T[JJ]*F
IF {MPOR(RJ,[]) 57,57,56
DREF=AMAX1{DREF,RMDP(5,MPJ)}
                L141-56
                EFIOREF-RHOP(6.8PJ)167.57.57
               COT+COSQINUIGO & CET+CE(NJ.4)
 57
                GO TO TO
```

SUBROUTINE JSTRESS

```
PRESSURE AND TENSILE STRENGTH IN POHILS
                                                                     CALL POREGITTES, CLUSS, PLUS, PS, PPS, CLUSS
                    L(6)=60
                    VADEM-VACDPIAC, MPJ)
                    TJ=0.
                    TUMO.

IF (TUJ) -NE. C.) 65-67

TUM(TENS(MJ.2)+(TENS(MJ.1)-TENS(PJ.2))+(GREF-RHC(PJ))/

(RMCP(5,PPJ)-RMC(PJ)))+F
                    COT-COSQUALINC1+COSQUALINC+L1-CCSCUALINC11+CDREF -RHOPINCIMPALL
                   /{BMCPINC+1,PPJ}-RHOPINC,MPJ}}
C1T=C1[MJ,NC+(C1(MJ,NC+1)-C1(MJ,NC))+(CREF -RHCPINC,MFJ)}
                          /{RMCFik6+1,PPJ)-EHCF(NC,NPJ))
                    L(41-67
                                                  ARTIFICIAL VISCOSITY
70
                    CU-APIALICU.O.)
                    L(9)-70
                    CJ=(0U+COT+CLT+C(JJ))+CU+(C(JJ)+BCLD)/2.
c
                                                DEVIATOR STRESS CALCULATICA
                   TF (ASR(MJ) .EC. 0) GC TC 78
GRHO=CIJJ+CCLC S CAYG=(CIJJ)+CCL0)/2.
SDO=SDJ S CCEF=P(P+DAHC/CAYG S PECh=2.+Phup
ASPM-ASP(MJ) S CT=.5+DTNH
GO TO (73,74,74,73,75) ASRP
                     CALL RELAX(SCJ.FJ3.YJ.SCG.DT.TSR(MJ.1).CRFG.CCEF.YACOM.A.JJ.
                         IMMEA, (LMIOY, IS, LMIRET
                    GG 1C #G
                    CALL BANDAL MESOU, PUB, VJ. SOC. DT. ORHC . SCEF . PECH, YADDP . N. JJ. TSREPJ. 11
                 1 TSR(NJ,2),TSR(PJ,3),TSR(NJ,4),TSR(PJ,5),TSR(NJ,4),NEM(JJ),
2 NETCJJ),NSRN)
                     23 01 03
                    CALL MARCHITATE THE THE THE TERM OF THE TERM OF THE TRANSPORT OF THE TRANS
                     $DJ-$DJ-2.9MUM-0E(JJ)-DC(U)/(E(JJ)+DC(E))
IF (AB$($DJ) .LT. YJ9F) 60 TO 60
                    100.(9+110130-101384+M0DA++448588811M1NA1M312-402
                    STRESS COMPUTATION

SCIJI=PEJJI+SDJ 8 REJJI=S(JJ)+C)

IFELSURESI_SE_MAMPR(2)) TO TO SC

MRITE(R#,5000)NC.PJ=PFJ=CJ-SDJ+F-TJ+(L(1)+1-1+10)
                                                                                                                                                                                                                                                   12/17
                                                                                                                                                                                                                                                   12/17
                    1206131-F208(5)+1
                  PORRAT(* JSTRESS NC++,11,4 NJ++,11,4 PFJ++,11,4 CJ=+,E10.3,

1 • $0J-+,E10.3,4 F-+,E1C.3,4TJ-+,E1C.2,4 L=+,(012)

FORRAT 120H STOP IN JSTRESS, N+14,4H, J-14,4H, D+E1C.3,

1 7H, TIME+E10.3)
    METLER
SOOD FORMAT(+
                1
 110
                  1
                     € MC
```

SUBROUTINE JSTRESS (concluded)

```
SUBROUTINE POREQST(EJ.DJ.PJ.PJ.PJ.CJ)
                   COMPUTES PRESSURE, BULK AND SHEAR MODILI, AND SCUND SPEED FOR
                   PORCLS PATERIAL
                   REAL MATL. MU. MUM. MLP
                                               MISCELLANECUS
                  COMMON CEFTCKS.DAYG.DELTIM.DOLD.DRHCHTEPIA.CIA.DTAH.DU.GX.ECLD.F.
FIRST,J.JCYCS.JIAIT.JFIA.JSMAX.JST/H.JTS.LSLELIIJ.MAXPRL.11.N.
COCCS.NPERN.POLC.RLAST.SLAST.SPAX.TF.FIPE.TJ.TS.ULAST.UCLC.
                        PLAST, XNC+, XCLO
ECUATION OF STATE - SOLID
                  COMMON CCSG(6.e).c(16.f).c2(6).ePELT(6.5).ECSTA(6).EQSTC(6).
L ECSTD(6).ECSTE(6).ECSTC(6).ECSTF(6).EQSTC(6).MATL(2.6).
MUL6).RNC(6).RHCS(6).TERS(6.3).YACZ(6).YO(6).JERC(6).MPOR(6.2).
                         P.PUP, ARTRLS
                                             EQUATION OF STATE - PCACUS
                   COPMCH AK(6), MUP(6), PCRA(4,3), PCRB(4,2), PCRC(4,3), RHOP(6,3),
                      YACCPE4.31. DREF . RHGPV. NP.NC
                   CIPENSICK L(10)
                  CO & 1=1,3
                   (C=0.0
                                                                                                                                                                                                                                          12/17
                  PI=PCRA(2,PPJ)
                   GC 10 30
                  CDENSITY LIES IN THE LAST PORCES REGION BEFORE SOLID)

CALL ECST(EJ,RHCPV,FJCIAT,PJ,CC)

(SIMPLIFY EXPAESSIONS AND COMPLTE PRESSURE ALONG THE CURVE)

PAF=PORK(NC,NPJ)+F & DA=NHCP(AC,CFJ)/TF

PCF=4.**PORCINC,NPJ)+FCN/(RHOPV-DN)+(RHCPV/DJ-1.)

P2=PAF-RHOPV/(RHOPV-DN)+(1.**-OA/DJ)+(PJCIAT-F8F+FCF)
20
                   ILUM DE LE S 12-DESENDE LES
                   1111-20
                                                COPPLIE BULK PRESSURE AND SELECT PINIMUM
                  RATIC-ARINI(1.,AMAX1((P2-P11/(PCR8(4,NPJ)-P11,C.))
MUM-(NLP(NJ)-EPU(PJ)-PUP(PJ)1068T/C)0F
                     THE DE LANGE LENS THE STEED TO SELECT A STATE OF THE STAT
                  L121-40
                   FJ-POLLE 1 GO TE 50
45
                                               STANE SPEEC CALCULATION
                   CSW-1004Rettun 1/1.50101-0CL011
                  IF ICSULEY. OLI STYAT
CO-CSQ+CJ+CJ S CJ+CSQ+CJ/CQ+L25+CV/CJ
                                                                                                                                                                                                                                         12/17
                   88'(LSUB(S)-68-PAXPR(S)) DD 70 93
NR178 (30-310) Johorschell-SC-CJ-DJ-62-PJ-62-POLLK-BULK-MUP-EREF.
| MHOPY-DDLD-TF-8CLD-CSC-.1111-3-13)
                   LSUMISING SUBIST + &
       **
                   S. A IF AGO
                   RETURA
                  CORRET (110 PORECST Jets.30 Rete.34 Pe222.60 SCet2.46 EJe21C.3,

6 AN BJ=10.3.60 CJ=210.3.30 Pe3210.3.50 ROD=2210.3/51.200=3210.3,

7 AN TP=210.3.60 ECLO=C10.3.50 CSC=210.3.30 L=313)
```

### SUBROUTINE POREOST (complete)

```
SUSROUTINE REZONE
        INCREASES CELL SIZES TO GIVE MORE UNIFORM DISTRIBUTION

* STARTS REZONING AT JREZON AND WORKS TOWARD JINIT
                DOFS GOT DISTURB LOCATION OF INTERFACES. JEDITS, OR SPALLS
        REAL MATE, MU, MUM, MUP, NEM, NE
REAL MASS, MOM, MASEAST, MOMEAST, MAGNEXT, MOMNERT
        INTEGER H. POROUS, RINTER, SOLID, SPALL
        INTEGER HC. HJOLD2
                     MISCELLANEOUS
C
        COMMON CEF,CKS,DAYG,DELTIM,DOLD,DRHO,DTM,4,DIN,DTNH,DU,DX,EOLD,F,
FIRSTAJ,JCYCS,JINIT,JFIN,JSMAX,JSTAR,JTS,LSUB(11),MAXPR(11),N,
CCYCS,NPERN,POLD,RLAST,SLAST,SMAX,TF,TIME,TJ,TS,ULAST,UGLD,
         XLAST, XNOW, XOLD
                    EQUATION OF STATE - SOLID
C
        COMMON COSQ(6,6),C1(6,6),C2(6),EMELT(3,5),EQSTA(6),EQSTC(6),
         EQSTD16), EQSTE16), EQSTG16), EQSTH16), EQSTN16), EQSTS16), MATL12, 6),
           MU(6),RHO(6),RHOS(6),TENS(6,3),YADD(6),YO(6),J9ND(6),NPOR(6,2),
          M, MUM, NMTRLS
       EQUATION OF STATE - POROUS

COMMON AK(6), MUP(6), PORA(4,3), PORB(4,3), PORC(4,3), RHOP(6,3),

1 YADDP(4,3), OREF, RHOPV, MP, NC

RADIATION DEPOSITION
        COMMON SS(300,5).SSTOP(5).START(5).SDURA.SSTOPM.NSPEC.SSJ.JSS
COORDINATE ARRAYS
        COMMON X(300),C(300),CHL(300),D(300),DHL(300),EHL(300),H(300,3)
           NEM(300), NET(300), P(300), Phil (300), R(300), S(300), SHL (300), Y(300),
       2 U13001, UHL [300], YHL [
                                            L4613001
        COMMON DH. DHLAST. DUH. FS., PH. RH. RHLAST. SH. SHLAST. UH. UHLAST. XH. XHLAST
                     IDENTIFIERS
        COMMON DISCPT(10), DENT, JEDIT(20), JREZON(15), NDATE(3), NEOIT, NEDTM,
NJEDIT, NR, NREZON, NSEPRÆT NSPALL, NTEDT, NTEX, NTR(15), TEDIT(50)
CONDITION ENDICE RS
        COMMON INF, LINTER, MIRROR, NORMAL, POROUS, RINTER, SOLID, SPALL
                     SPALL AND RELAXATION
        COMMON NER(6), TER(6,6), USP(50), XER(50), LRLX. NICK, NEMH, NETH
        GIMENSION CC(2%, +DXX(6), EC(20), HC(20,3), MASS(21), MOM(20,2), L PC(20), RS(21), SC(20), XC(20), YC(20), ANEM(20), ANET(20)
        DIMENSION AM(20), AT(20), NEWJED(20)
        CALL SECOND ( XNOW)
         INTEL ( INC. SAN) NO. SAN LONINE NO. SAN LING.
        IF (JREZON(NREZON) .LE. JINIT) GO TO 900 IF (TIME+DTNH .LT. SSTOPM) GO TO 900
        IF ITHE-.5*DINH .GT. SSTOPH) GO TO 2
TIME =TIME+BINH $ SEGRM=G.
DO 1 3=JINIT, JFIN
         EHL(JI=EHL(J)+SSCALH(J)
        TIME=TIME=DTNH $ SOURM=1.
JLAST=JRFZON(NREZON!-1 $ M=0 $ NJ=1
               SECTION 1 LOCATE JREZON WITH RESPECT TO MATERIAL AND JEDITS
        IF (JREZON(MREZON)~JBND(M)-216,5,3
         JLAST=JLAS:-1
         MASLAST=ZHL (JLAST)
        MOMLAST=0.5+MASLAST+U(JLAST)
        RSLAST=.5*MASLAST*(RIJLAST)+RIJLAST+11) & TLAST=TIJLAST)
        DO 10 1=1.NJEDIT
         NEWJED(1)=JEDIT(!)
              SET JOLD, THE OLD COORDINATE VALUE, AND JNEW, THE NEW VALUE
REZONING OCCURS FOR CELLS BETWEEN JOLD AND JLAST. MIDCELL
              QUANTITIES ARE SET FOR JEAST-1 WHILE COORDINATE QUANTITIES ARE
        ** SET FOR JLAST.
JOLD=JNEW JLAST-1 $ NCEL=NPART=0
```

SUE REZONE

```
WRITE (18,5000) LOC.JOLD.JNEW.JLAST.M.NJ.MCEL.NPART

** SET DX (CELL DIMENSION) AND XN (COORD +D LEFT OF NEW CELL) FOR

** FIRST GROUP OF CELLS TO BE REZONED

XN=X(JLAST-1) & DX=X(JLAST)=:(N

M*=M-1 & DXX(M)=DX

IF (M .EQ. 1) GO TO 13
                                                                                                                               PRINT
C
          DXX(1)=DXX(M)+SQRT(EQSYC(1)+RHOS(M)/(EQSYC(M)+RHOS(1)))
          WRITE (18,5015) (1,0XX(1), (=1,4)
13
          ** SAVE NEM AND MET AT JEDITS FOR RADIATION GEPOSITION PROBLEMS IF (NSPEC .EQ. 0) GO TO 50 DO 30 I-1.NJEDIT
          AM(I)=NEM(JE)
          ATILI=MET(JE)
30
                     SECTION 2 - FIND REZONABLE SET OF CELLS
                    TERMINATION OF REZONABLE SET OF CELLS AT AN INTERFACE (PART 1)
          IF (M-1) 790,130,52
IF (JOLD-JBND(N-1)-1) 790,60,130
NPART=1 $ HJOLD2=H(JOLD,2) $ CO TO 900
52
60
          JLAST=JOLD-1
100
           ** RETURN WITH JNEW SET TO JBND(M-1)
CHECK WHETHER JEDITS COINCIDE WITH JBND(M-1) OR JBND(M-1)+1
          DO 103 NJ=1, NJEDIT
          IF (JOLD .NE. JEDIT(NJ)) GO 10 103
NEWJED(NJ)*JNEW+1
          GC TO 105
          CONTINUE
          ON 108 NJ=1, MJEDIT
IF (JOLD-1 .NE. JEDIT(NJ)) GD TO 108
NEWJED(NJ)=JNEW
105
          GO TO 125
108
125
          M=M-1 $ JBND(M)=JNEW $ X(JNEW)=X(JNEW+1;
M=M-1 $ JBND(M)=JNEW $ TLAST=T(JOLD-1) $ JNEW=JNEW-1
           JOLD+JGLD-2 8 XN=X(JOLD)
          MRTTE (18,5000) (OC,JOLD,JNEW,JEAST,M,NJ,NTEL,NPART GO TO 50
                                                                                                                               PRINT
          TERMINATION AT A SPALL WITHIN MATERIAL (PART 5)
IF (IMIJGLD,2) .EQ. SPALL) .UR. (MIJGLD,2) .EQ. NGRMAL)) 155,132
NPART=5 & NSP=H(JOLD,2) & XSAVE=X(JOLD) & USAVE=U(JOLD)
X(JOLD)=XSP(NSP) & U(JOLD)=USP(NSP) & GO TO 500
132
          ALJULUT-ASPINSPI B U(JULUTHUSP(NSP) $ G0 TO 500

JLAST*JOLD $ USP(NSP)=UIJNEW+1)

XSP(NSP)=X[JNEW+1] $ X(JULUT*XSAVE $ U(JOLOT*USAVE

RETURN WITH JNEW =COORD TO LEFT OF SPALL, JOLO*COORD AY SPALL

M(JNEW+1,2)*NSP
          DG 142 NJ=1,NJEDIT
IF (JOLD .ME. JEDIT(NJ)) GG TO 142
NEWJED(NJ)=JNEW
          GO TO 145
          JOLD=J060-1
          WRITE [18:5000] LOC, JOLD, JNEW, JLAST, M, NJ, NCEL, NPART
                    TERMINATION AT INITIAL BOUNDARY (PART 2)
         IF(JOLD-JINIT) 790,160,255

NP3RT=2 $ GD TC 500

DO 205 NJ=1.NJEDIT

IF (JOLD .NE. JEDIT(NJ)) GO TO 205

NEWED(NJ) #JMEW+1
 155
                                                                                                                                12/17
160
```

```
CONTINUE
205
               TERMINATION OF REZONABLE SET OF CELLS AT A JEDIT (PART 3)
        IF (NJEDIT .LE. 0) GO fo 355
DO 264 NJ#1, NJEDIT
        IF (JOLD .NE. JEDITINJI) GO TO 264
        MPART=3
        60 TO 500
264
        CONTINUE
        60 TO 355
        JOLD=JOLD-1
300
        NEWJEDINJ)=JNEW+1

** RETURN WITH JORG AT THE JEDIT, JNEW TO THE LEFT OF JEDIT. MIDCELL

** QUANTITIES HAVE REFN RESET UP TO JNEW+1. COORDINATE QUANTITIES
           UP TO JNEW+2
        LOC=300
        WRITE (18,5000) LOC. JOLD, JNEW-JLAST, M. NJ. NCEL, NPART
                                                                                               PRINT
        GO TO 50
Ç
               TERMINATION WHEN NUMBER OF REZOMABLE DED CEL " IS 20 (PART 4)
        IF ((x(JLAST)-x(JULD))/DX((M)-1A-) 420,360,360
NPART=4 3 GO TO 500
355
360
400
        JOLD# JOLD- 1
        LOC=400
             RETURN WITH JOLD AT PREVIOUS LOCATION, JNEW SET AT COORDINATE TO THE
        ## REFT. MIDGELL QUANTITIES HAVE BEEN RESET UP TO JNEM+1, COGROENATE
## QUANTITIES UP TO JNEM+2
## (19,5000) LOC, JOLD+JNEW+JL&ST, M+NJ+NGEL+NPART PRINT
GO TO 50
                                                                                               PRINT
        JOLD-JOLD-1 $ GO TO 50
420
                SECTION 3 - COMPUTE NEW CELL COORDINATES AND PROPERTIES
500
        NO=0
        HRTTE (18,5000) LOC.JOLD.J.VEW.JLAST.M.NJ.NCEL.NPART
NCEL-MAXITTXIJLAST)-XLJOLDTY/DXXLM)+.65,1.)
IF (INCSL-1)+(NQ-1) .EQ. 0) GO TO 610
                                                                                               PRINT
510
        CHECK WHETHER REGION OF LARGE CELLS LIES TO LEFT DEMINEDEX(M) & JLASTPEJLAST-L
601
        LOC*601
        WRITE (18,5000) LOC.JOLD, JNEW, JLAST, M. NJ, NCEL, NPART
                                                                                               PRINT
        DO 603 JX=JOED, JLASTS
DELX=X(JX+18-X(JX)
        IF (DELX-DXMIN) 602,663,603
        DXMIN=DELX & JXMIN=JX
602
603
        CONTINUE
        IF (DXMIN-G.8+DXX(M)) 604,750,750
        JX=JX@IN+1
604
        605
608
        LOC=604
        WRITE (18,5000) LOC.JOLO.JNEW.JLAST.M.NJ.NCEL.NPART
NQ=L & GD TQ 510
                                                                                               PRINT
C
                    REGIN COMPUTATIONS FOR NEW COORDINATES
610
        NCEL=MINO(JLAST-JOLD,NCFL)
        DX=[X(JLAST]-X(JOLD))/NCEL
XSTART=X(JLAST) & XN=XSTART-DX
        ** XN IS NEW COORDINATE LOCATION
** DX IS NEW CELL DIMENSION
        RS(1)=RSLAST
        MOM (1,1)=MOMLAST
        MASSILI=MASLAST
        LOC=510
```

```
WRITE (18,5002) LOC.NCEL,XSTART,DX,XN,RSLAST.MASLAST,MOMLAS) HRITE (18,5610)
         MP=NPOR(4,21
        DO 650 1=1.NCEL
        RS([+1]=MASS[[+1]=AMAVG=AMSLP1=AMSLP2=ENGY=CS=PS=SX=YS=O.
        HC(I.X)=SOLID
        ANEMS=ANETS=O.
        HC (1,3)=2
         IF (JLAST .LT. 1) GO TO 625
        XEND=AMAXI(X(JLAST)+XN)

IF (XSTART-XEND) 621+621+619

DHASS=ZHL(JLAST)+(XSTART-XEND)/(X(JLAST+1)-X(JLAST))
        MASS(1+1)=MASS(1+1)+DMASS
        UJ=U(JLAST)
        DUOLD=UIJLAST+11-UJ
        DXOLD=X(JLAST+1)-X(JLAST)
        XS1=XSTART+XEND-XN-XN-DX
        XSZ=XSTART-XEND
AMAVG=_5+DMASS+(UJ+DUDLD+({XSTART+XEND}/2.-X{JLAST}}/DXOLD}+AMAVG
        ###YUR_5=PD##ASS/DX+1.25+DUOLD/DXOLD+XS2+(3.*XS1+XS2)
##SLF1=B##$S/DX+1.25+DUOLD/DXOLD+XS2+(3.*XS1+XS2)
##SLF1=B##$UJ+DUOLD+(XEND-X(JLAST))/DXOLD)+XS1) + AMSLF1
##SLF2=#1.5+DM#CS/DX+XS1 + AMSLF2
ENGY=FNGY+DM#SS+EHL(JL#ST)
        RS(1+1)=RS(1+1)+DMASS*(R(JLAST)+(R(JLAST+1)-R(JLAST))*((XSTART+
        XFND1/2.-X[JLAST])/(X(JLAST+1)-X[JLAST]))
PS=PS+DMASS+PHL(JLAST)
         SX=SX+DMASS+SHL(JLAST)
         YS=YS+DMASS=YHL(JLAST)
        CS=CS+DMASS+CHL(JLAST)
ANEHS=ANEHS+DHASS+NEH(JLAST)
         ANETS=ANETS+DMASS=NET(JLAST)
        XSTART=XEND
        TF (H(JLAST,1) .FQ. POROUS) HC([,1)=POROUS
HC([,3)=MAXO(H(JLAST,3),HC([,3))
        IF (XEND .LE. XN) GO TO 625

JLAST-JLAST-1 $ GO TO 615

XC(1)=XN $ DC=MASS(1+1)/DX $ EC(1)=ENGY/MASS(1+1)/OX

YC(1)=XS/MASS(1+1) $ SC(1)=SX/MASS(1+1) $ PC(1)=PS/MASS(1+1)
621
                                                                   $ EC(1)=ENGY/MASS(1+1)
        CC(1)=CS/MASS(1+1)
AMSLP=AMSLP1+2.#AMAVG+AMSLP2/MASS(1+1)
         MOH( I . 2) = AMAVG+AMSL P
         MOM(1+1+1) = AMAVG-AMSLP
        ANEM(I)=ANEMS/MASS(I+11 & ANET(
IF (HC(I+1) .EQ. SOLID) GO TG 643
                                                ANET(1)=ANETS/MASS(1+1)
        HC(1)1)=POROUS

TF (EC(1) .GT. EMELT(M,1)) GO TO 640

RHOPY=FHELT(EC(1),M)+(RHOP(5,MP)-RHOS(M))+RHOS(M)/(1.+EC(1)
          EQSTAIMIT
        IF (DC .LT. RHOPY) GO TO 643
HC(1:1)=SOLID 3 GO TO 643
IF (PC(1) .GT. O.) GO TO 643
640
         PC ( | ) = SC ( | ) = R S ( | + | ) = 0.
643
        K=JNEW+1-1
        100=643
         WRITE (18,5004) LOC, K, XC(1)+DC, MOM(1+2), MOM(1+1+1), EC(1), RS(1+1).
           PCIII,SCIII, YCIII, MASSII+11, HCII,11
650
        XM=AMAX1:XM-DX,X(JOLD)?
        T(JNEW+1)=TLAST
                                                                   $ DHL(J)=MASS((+1)/DK
$ SHL(J)=SC(()
         J= JNFW+1-1
                                   $ CHLIJI=CCII)
                                   B PHL(J)=PC(T)
B SHL(J)=MASS(I+1)
         EHL (3) = EC (1)
         YHL(J)=YC(I)
                                                                   $ HIJ, 11=HC(1.1)
         NET(J) -ARET(II
                                        NEM(J)=ANFH(I)
        H(J,3)=HC(1,3)
670
        CONTINUE
         MOMEASY - MOMENCEL +1 . LT
```

```
MASLAST=MASS(NCEL+1)
                          S TLAST=T(JOLD)
 460
                           RIJI=RSINCEL+11/MASSINCEL+11
885
                           TIJ1=TLAST
                          U(J)=2.4NOMLAST/MASLAST
MOMLAST=MASLAST=RSLAST=0.
700
                          CONTINUE
                          LOC=700
                          WRITE (18,5000) LOC.JOLD.JNEW.JLAST.N.NJ.NCEL.NPART
SET JNEW AND JLAST IN PREPARATION FOR THE NEXT ZONE CALCUL HIOMS
                                                                                                                                                                                                                                                                                                                     PRINT
C
                          THE JE STATE OF THE STATE OF TH
C
                           GO TO (100,200,300,400,140) NPART
                                                                RENUMBER CELLS WITHOUT REZONING
 750
                          TIJMEW+11=TLAST
                           LOC=750
                           TLAST=T(JOLD)
                          RSLAST=.502HL(JOLD)+(R(JOLD)+R(JOLD+1))
WRITE (18,5~3)
                           JLAST=JLAST-L
 752
                                                                                                                            $ DHL(JNEW)=DHL(JLAST)
                         ### CANAGE AND THE CONTROL OF THE CO
                           MASNEXT-ZHL ( JLAST )
                           U(JNEH+1)=(2.*MOMLAST+MASNEXT+U(JLAST+1))/(MASLAST+MASHEXT)
                           MASLAST-MASHEXT
                            MONLAST=.5+MASLAST+U(JLAST)
                           RIJNEW+11=RIJLAST+11
                                                                                                                        $ X(JNEW)=X(JLAST)
$ NET(JNEW)=NET(JLAST)
                           TIJNEW)=TIJLAST)
                           NEM ( JNEW) = NEM ( JLAST )
                           HIJNEW, 3)-HIJLAST, 3)
                                                                                                                                         JNEM= JNEM- J
                          TEJNEW SOUTH STATE OF THE LAST RENUMBERED CELL. JLAST=JOLD, THE LAST PO OLD CORROLLER.
  755
                          CONTINUE
                           LOC=755
                           WRITE (18,5000) LOC, JOLD, JNEW, JLAST, M, NJ, NCEL, NPART
GO TO (760,760,300,400,755) NPART
                                                                                                                                                                                                                                                                                                                      PRINT
 760
                           RIJNEH+1)=RIJLAST)
                           U(JNEW+1)=2. #MOMLAST/MASLAST
T(JNEW+1)=T(JLAST)
                           MOMLAST=MASLAST=RSLAST=0.
                           LOC=760
                           WRITE (18,5000) LOC.JOLD,JNEW,JLAST,N.NJ.NCEL,NPART
GO TO (100,200,300,400,140) NPART
                                                                                                                                                                                                                                                                                                                      PRINT
 C
                                                    ERROR MESSAGE
                          WRITE (18,1000) NPART, JOLD, JNEW, JLAST, NJ., JEDIT(NJ), M, JENO(M) CALL EDDUMP 8 LSUB(7)=1 8 CALL SCRIBE 8 STOP
  790
                                                                                                                                                                                                                                                                                                                      PRINT
 C
                                                    ENDING ROUTINE - INTERFACE AND BOUNDARY ADJUSTMENTS
  800
                           JIMIT-JMEN+1 $ H(JINIT,2)=H(JOLD,2)
                          JINITAJNENT S HIJINITA SHIJOLDAZI

IF (HIJINITAZI) EQ. SPALLI RIJINITI = SIJINITI = PIJINITI = 0.

DO 810 J=JINITAJPIN

UML[J]=.S=(U[J]+U[J+1])

SIJ+1]=.S=(SHL[J]+SHL[J+1])
                           P(J+1)=.5*(PHL(J)+PHL(J+1))
 810
                           C(J)=CHL(J)
                          DO 820 M=1, MMTRLS
JG=JBNO(M) S M(JB, Z)=LINTER
S(JB)=SML(JB=1)
```

```
$(JB41)=$HL(JB+1)
P(JB)=PHL(JB-1)
                 P(J8+1)=PHL(J8+1)
                IF (M(JB01,2) .EQ. SPALL) 815.816
R(J8+1)=R(J8)=0. $ GO TO 820
R(J8+1)=R(J8)=.5*(R(J5+1)+R(J8))
816
                 U(JB+1)=U(JB)=(U(JB)07HL(JB-1)+U(JB+1)+ZHL(JB+1))/(ZHL(JB-1)+
                CONTINUE
820
                CONTINUE
DO 825 I=1,NJEDIT
JEDITIII=NENJEDII
IF (NSPEC .EQ. 0) GO TO 840
DO 830 I=1,NJEDIT
JE=JEDITII

                 MEM(JE)=AM(I)
MET(JE)=AT(I)
630
                CALL EDIT
CALL EDOUMP
CONTINUE
CALL SECOND(TWIX) S DUR-TWIX-XNOW
WRITE (18,5010) JINIT, DUR
840
900
                 RETURN
RETURN

FORMAT (25M ERROR IN REZONE, NPART=13.6H JOLD=13.6H JNEW=13,

1 7H JLAST=13.4H NJ=13.11H JED[T(H)]=13.3H M=13.9H JBND(M)=13)

5000 FORMAT (13H REZONE, LOC=13.7H, JDLD=13.7H, JNEW=13.8H, JLAST=13.

1 4H, M=13.5H, NJ=13.7H, NCEL=13.6H, NPART=13)

5002 FORMAT (13H REZONE, LOC=13.7H, NCEL=13.9H, XSTART=E10.3.5H, DX=,

1 E10.3.5H, XM=E10.3.9H, RSLAST=E10.3.10H, MASLAST=E10.3.
             1 E10.3,5% XM=E10.3,5%; RSLAST=E10.3,10% MASLAST=E10.3,
2 10% NOMLAST=E10.3)
FORMAT(2:5,10E10.3,3X,R1)
FORMAT (2:5,1P10E10.3,3X,R1)
FORMAT(19MOEND REZONE, JINIT=13.17%, TIME IN REZONE=E10.3,5% SECS)
FORMAT(2:M REZONE, DXX VALUES ,6(18,F9.6))
FORMAT (1:20M LOC J XC DC MOM(1,2) MOM(1+1,1)

1 EC RS(1+1) PC SC YC MASS(1+1) HC(1,1)
5003
5004
 5010
5015
5610
                                                                                                                                            U(1+1)
                                                                        J
SHL
                                                                                                   X YHL
5750 FORMAT (120H LOC
1R(1+1) PHL
                                                                                                                           DHL
                                                                                                                                                                                 FML H(1.1)
                END
```

385+

SUBROUTINE REZONE (concluded)

```
FUNCTION SSCALIJS!
        (INCLUDES ENTRY SSCALH) CG. PUTES RADIANT ENERGY FOR DEPOSITION IN EACH CELL DURING THE
         HYDRODYNAMIC COMPUTATIONS
             * SSCAL COMPUTES FOR COORDINATE POINTS
* SSCALH COMPUTES FOR HALFSTEP POINTS AND INITIALIZES ENERGY IN
                  HEW ZONES
        REAL MATL, MU, NUM, MUP, NEM, NET
        INTEGER H
HISCELLANEOUS
       COMMON CEF,CKS,DAYG,DELTIM,DDLD,DRHO,DTMIN,DTN,DTNH,DU,DX,EDLD,F,
1 FIRST,J,JCYCS,JINIT,JFIN,JSMAX,JSTAR,JTS,LSUB(11),MAXPR(11),N,
2 NCYCS,NPERN,POLD,RLAST,SLAST,SMAX,TF,TIME,TJ,TS,ULAST,UDLD,
           XLAST . XNOW . XOLD
        EQUATION OF STATE - SOLID

COMMON COSQ(6,6),C1(6,6),C2(6),EMELT(6,5),EQSTA(6),EQSTC(6),

EQSTD(6),EQSTE(6),EQSTG(6),EQSTH(6),EQSTN(6),EQSTS(6),MATL(2,6),
C
           MU(6),RHO(6),RHOS(6),TENS(6,3),YADD(6),YO(6),JBND(6),NPOR(6,2),
       3 M. MUH. NM TRLS
       EQUATION OF STATE - POROUS

COMMON AK(4):NUP(6):PORA(4.3):PORB(4.3):PORC(4.3):RHOP(6.3):

YADDP(4.3):DREF:RHOPV:MP:NC

RADIATION DEPOSITION
        COMMON SS(300,5), SSTOP(5), START(5), SDURM, SSTOPM, NSPEC, SSJ, JSS
                     COORDINATE ARRAYS
        CDMMON X(300),C(300),CHL(300),D(300),DHL(300),EHL(300),H(300,3),
    NEM(300),NET(300),P(300),PhL(300),R(300),S(300),SHL(300),T(300),
       2 U(300), UHL(300), YHL(300), ZHL(300)
         DIMENSION L(10)
         SSCAL *O.
        IF (NSPEC+(SDURM-3.) .EQ. O.) RETURN
IF (TIME-.5+DTNH .GT. SSTOPM) RETURN
         00 1 1=1,10
         L(1)=0
         1(1)=5
                     ENERGY ADDITION AT COORDINATE POIRTS
10
        IF ((TIME-START(1))*(TIME-.5+DTNH-SSTOP(1))) 12,20,20
SSCAL=SSCAL+.5+SS(JS,1)*AMIN1(AMIN1(.5+DTNH,TIME-SYART(1)),
12
          SSTOP(1)-TIME+.5*DTNH)
50
        CONTINUE
        GO TO 70
                    ENTRY FOR HALFSTEP CALCULATIONS
        ENTRY SSCALH
                                                                                                       ENTRY
        SSCAL=0.
IF (NSPEC .EQ.O) RETURN
IF (JS .GT. JSS) GO TO 50
IF (SDURM .EQ. 1.) RETURN
IF (TIME-DTNH-.5+DTN .GT. SSTOPM) RETURN
         SSCAL + O
        00 42 1=1,10
       L(1)=0
                    ENERGY ADDITION IN ACTIVE ZONES - HALF STEP
C
        DO 45 1-1 . NSPEC
         IF (ITTME-.5*DTNH-START(1))*ITTME-DTNH-.5*DTN-SSTOP(1))) 46,48,48
        SSCAL +SSCAL +SS(JS+1)+(AMINL(SSTOP(1),TIME-.5+DTNH)-
           AMAX (START( I) . TIME-DTNH-.5+DTN)
4,8
        CONTINUE
        GO TO 70
                     ENERGY ADDITION FOR NEW ZONES
50
        1121=50
        DO 60 I*1; MSPEC IF (TIME-.5*OTHH .LT. START(I)) GO TO 60
         SSCAL=SSCAL+SS(JS+1)+(AMIN1(SSTOP(I),TIME-+5+OTNH)-START(I))
60
70
        CONT INUF
         IF (LSUBIO) .GE. MAXPRIO) GO TO 75
         WRITE (18,5000) SSCAL-L(1)+L(2)
        LSUB(6) = LSUB(6) + 1
         RETURN
5000
        FORMAT(5x, GHSSCAL > E10. 4.3H L=213)
        END
```

```
SUBROUTINE RELAXISO, ICON, YCLD, SDG, DT, TRLX, CRO, COEF, YAD, N. J. TY.
     LYNCT, INSRI
THIS VERSION CONTAINS STRESS RELAXATION PODELS 1 AND 4
      DIMENSION JK (20)
      KKK#O
       ICOR-ICON $ YNEB-YGLD $ L=0
      IN-PAXO(2, INSR)/2
      IF (ICON-EG.2)18.1
IN HAL CONDITION GUTSIDE OF ELASTIC ZONE
      L 1 $ _0 TO (4,2) IN APY=EXP(-DT/1Y)
      KKK=KKK+1
      JK (KKK)=2
      YNEh=YNCT+(YCLD-YNCT)+XPY+AES(COEF)+TY/CT+(1.-XPY)+.5+(1.+SIGN(1..
     1CRC) = SIGN(1., SDO))
      YAVG = (YMEW+YOLD)/2.
Y'TAR = SIGN(YOLC, SOC)
                               $ YSTAR = SIGNIYAVG.SCO) $ GO TC 5
       KKK=KKK+1
       JK ( XK) )=4
      XPO=EXP(-OT; TRLX)
      RKK=KKK+1
       JK (KKK)=5
      C
       *********
       ~ * ( KKK ) ~ 6
      GO TO (5.7) IN
       YNEL=YNOT+(YGLD-YACT)*EXP(-TC/TY)
      KKK=KK. >1
       JK (KKK) =7
       YAVE = (YMEH+YCLC)/2. $ YSTAR = SIGA(YAVC, SOC)
       XPC = EXPI-.C/TRLX)
      KKK=KKK+1
       JK ( KKK ) = 9
       SD=YSTAR+(SDC-YSTAR)+XPC+COEF+TRLX/CT+(1.-XPC)+CGEFC+(D1-TC)/DT
      C
10
C
12
      TK = (SD SIGN(YNEL,SDC))/(SD-SIGN(YNEW,SCC))+(CT-TC) $L=4
      KKK-KKK+1
       JK { KKK } = 12
      GO TO 125.13) IN
XPY-EXP(-TK/TY)
                          S YNOLC=YNEL
13
       YNE b= YNGI+(YNEh-YACT)+XPY+ ABS(CCEF)+TY/CT+(1.-XPY)
      YAVG = (YNEW+'NGLC)/2. 8 YSTAR = SIGA(YAVG,SD)
NOW CONSIDER INITIAL CONDITIONS INSIDE ELASTIC ZONE
                                                              $ GE TC 25
      SD =SCC + CCEF
      KKK=KKK+1
       JK (XKK)=18
      CHECK IF DEVIATOR CHOSSES ZONE BOUNDARY
C
       IF (ABSISD).GT.YCLC)19,35
      C
19
       KKK-KKK+1
       JK ( KKK ) = 19
       GO TO 125.20)!M
THE B-YNGT + ABS (LCEF) + TY/DT+(1.-EXP(-TK/TY))
20
       KKK-KKK+1
       JK {KKK}=20
   YAMG = (YREN-YADT)/2. $ YSTAR = SIGA(YANG,SD)
25 ICCN = 2 - IPIX(SIGA(1.,SD))
       KKK-KKK+1
       JK(KKK)=25
SD = YSYAR +CCEF+TRLX/DT+(1-E>P(-TK/TRL>))
CO TO (31, 35) IN
30
       RECALCULATE VIELD STRENGTH TG ACCOUNT FOR STRAIN FARCENING VNEW - AMINICABS(SD), VGLD+YAD+ABS(GRC))
       I+XX=KXX+1
```

SUBROUTINE RELAX

SUBROUTINE RELAX concluded)

```
SUBRCUTINE BANCRLX(SC.ECON, YNCT, SEC.ET, URC.CCEF.FECH-YAC.A., J.T1.T2
1.BEE.YM.GEE.EPS, AP.AT.IASR)
THIS VERSICA CONTAINS BOTH BANC AND GILMAN RELAXATICH PEOELS
FOLLOWING TABLE GIVES CORPESPEACEFCE OF COMMON. BANCRLX VARIABLES
COPICA NS. TSR(1) (2) (2) (4) (5) (6) NEP AET
BANC 2 T1 T2 BEE VM GEE EPS NM NT
GILMAN 3 CEE PHI BEE VM BAPC - NM GAP
NOTE. MEDH-MUN. YAO-YAOD, INSF-NSR
NEM AND NET ARE MORILE AND TOTAL DISLOCATIONS
GAM IS PLASTIC SHEAR STRAIN
JK IS A PATH INDICATOR
                       JK IS A PATH INDICATOR
          CIMEASICA JK(20)
          REAL MECH, MM, NT, NMC . NTO
          KKK-0
          ICCR = ICQN & YNGTG=YNGT
ATC = AT & NPG = AP & SG=SDC
          ATT=4
          L=0
                   S ENT=FLOAT(NIT) SIT=C
          SIGHA = SIGN(1.,SEC)
IF(ICON.EQ.2)1,10
           INITIAL CONDITIONS INSIDE ELASTIC ZONE
          SD=SCO+COEF
          KKK=KKK+1
           JK(KKK;+1
          1F (A85(50).GT.YNOT)2,35
          CEVIATOR LEAVES ELISTIC ZONE. CALCULATE RELADATION
      2 L = 1
          KKK=KKK+1
          JK (KKK ) = 2
          JKERKE1=2

S = .5 = (ABSISDC+CLEF)-YNCT)

DELT = (SC-SIGN(YNCT,COEF))/(SC-SDC)>DT

SIGNN= SIGN(1.,CCEF)

ENT=1. 8 SC=SCO
          60 TO 4C
INITIAL CONCITION OUTSIDE OF ELASTIC ICAE
£
    10 f=5
           3K (KKK) = 10
          IT=IT+1 $SDI=SC+CCEF/(2.0ENT)
S=ABS(SCI)=YNOT $ OELT=DT/ENT
IF(S.LE.O.)10.11
          AVERAGE DEVIATOR REPAIRS OUTSIDE ELASTIC ICHE. CALCULATE PELAXATICN
C
     11 1=3
           JK (KKR)-11
IF(SIGNA-EG-SIGN(1-,SCI))e0,17
     13 IF(ABS(SD).GT.THOT)14,16
     14 1=4
          KEK-KEK+1
           AF-THEND NL
     DEVI-TOR REPAIRS CUTSIDE ELASTIC ZCAE SFTER RELAXATION 15 Les
           IF ($ 5 GM $ .. SE) . EC. SIGM) 15.14
C
           【《京湖山水湖南 4】
           J# ( ### ) = 15
          IFERT.EC.RITE 30,10
          $0-501-COEF/(2.0ENT)
lè
          REE-EEF +1
           Af - (XXX) - 14
          L-4
17
          KKK-KKK+1
           SKIKKK1-17
     DEVIATOR REENTERS ELASTIC ZCHE. RECALCULATE RELAZATION
18 5 - .50(ABS(SD)-VMOT) SYSTAR - SIGN(YACT-SEC)
          KKK-KKK+1
           JE ( KKK ) = 14
          DL_.={YSTAR-SD}/COEF=BT
GC TC 46
SD=SG+COEF/ENT+FLOAT(NIT-IT)
           RER-KEK+1
           J# ( KKE ) = 19
     # (ABS(SD).GT.YNCT)21,20
20 (CCA = 2
```

SUBROUTINE BANDRUX

```
KKK=KKK+1
                             JK ( KKK ) = 20
                             60 10 35
                       DEVIATOR CROSSES EVER TO EPPCSITE SICE OF ELASTIC ZONE. RECALCULATE RELAXIN
IF (SIGNIL., SC). EG. SIGHN) GC TO 30
21
                             KKK-KKK+1
                            JR(RR)=21
SIGHR = -SIGHN 8 L=7
GELT = (SD+YSTAR)/COEF+DT
                              1 = .50(ABS(SO)-YAGT)
                            S0-SC-CCEF & (
                                                                                     S ENT-1.
40
                              KKK=KKK+1
                              JE(KKK)=60
                            REK=EKE+1
                              JK (KKK)=63
                            GO TO 199,71,7233ASR
PERFORM RELAXATION CALCULATIONS - BARB POCEL
                              TP-NT
                              本在在一部本社+ ?
                             UKIKKKJATI
NT-NT-1EPS&GFEDSOINT-NM)-1./T2DNMOXPC}#BELT
                              MM=MM=(GEF+S+(TP-MM3-(1./T3+L./T2;+KM+MPLI+CELT
                            GC TC 74
PENFORM RELAXATION GALCULATIONS - GILPAN PCDEL
                             CEE-TI S PRINTE S BAMONGEE S CARRGARCHAY REROURES
 72
                               JKIFAK)-72
                               PHONENCALL. + CEEPGANIAEXPI-PI TACAPI
 74
                                SD-30+CCEF/ENT-PECWANPOVPOXFCOCELTOSICON
                               JAK 4"X+!
                              JR (RRF /- 74
GG TC (25, 19, 19, 19, 19, 25)
 75
              25 ICCA - 2-IFIX(SIGN(1.,521)
                                I+XXX=KKK+1
                                4K (KKK 1=25
                RECALCULATE VIELD STRENGTH IN CASES OF STRAIL HARDENING 36 YNOT - AMINITARSESCH, VACTOVACOALSTERCH
 C
                                美国家的复数形式主
                                UKERKA-30
IF (FOREG.ABSEST): 32.35
                32 ICON+7 B L+ E+7C-
REK+ERE+E
                                ACERS#3+32
                 35 CONTINUE
                                JR《张军报》235
                              LREASHING COLUMN COLUMN
                                CO TO TRELIEF ISSUEDS
PRINT PLOCUETY STATES
PRINT $10.5.MC.APC.UNU.De.25
  10
                                GC TC (40,48,473 INSC
MRINT E00,majje acon acoma soc, se come amace appliance at
                              GC TC TO
PRINT 101.M.J.L.ICON.ICON.SDC, SC. EF.TACT-GASC.GAP.AP
EZ ILAN
                                FRINT 199, INSR.L
             PORMATIZE MANATAL THE RANDRESS THE PROMET STATES OF THE STANDRESS TO THE STANDRESS TO THE STANDRESS THE PROMET TO THE STANDRESS THE PROMETERS OF THE PR
                           2410.3,1m-,E10.31
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SUBROUTINE BANDREY (con.inced)

SUBROUTINE BANDRLX (concluded)

```
SUBROUTINE BAUSCHE 41.SDC.SD.Y.YAGT.YAG.CRC.CCEFL.XPG.XP.MUL.N.J.
       REAL MU, MUL, MUU
        DIMENSION JL 1201
        KKK=0
        DO 90 K=1,20
90
        JL (X)=0
       DYMYAD SIOR=I SYOR=Y SCOEF=CCEFL SEMU=CY SSCAY=O. SMU=MUL
CCEFU=CCEF0#UU/MU
        IF(I.LE.2) GG TQ 3
DY-YADF-YAD SPU-FUL-PUU S CGEF-CGEFU
        KKK=KKK+1
        JE (KKK)=0
        SD= SDC+CGEF
        KKK=KKK+1
        JL (KKK)=3
        GO TO (10,5,10,18,59,18) [
     IFIXPO.GE.1.E236.1
DETERMINE EFFECTIVE SHEAR MCDULUS AND STRESS DEVIATOR FOR INITIAL LOADING
        EMU=MU+(DY*RG-MU)+(ABS(SCC/Y))+*xFC
        KKK BEKK + 1
        JL [KKK]=1
        SDAY=AMINICY, SDC+CGEF=EMU/ML/2.1
       EMU=NU+IDY*RC-NU;*(ABS(SCAV/Y))**XFC $ SC=SCC+CGEF*EMU/NU
IF(SD-GE-Y)2,7
IF (CRO-LT-0-)8,30
        PL=PLE=PUU SDY=YADF *YAD SCCEF=CCEFU SCO TO 20
        Y# AMINITABS(50), Y+150-Y1/150-SDC)+DY+A85(CRC)) $5C=51CA(Y,5C) $1=1
        KKK=KKK+1
        JL (KKK) #2
      JL(KKK)=2
GO TO 30
STRAIN MARDERING PATH
IF(ISIGN(1).IFIX(ORC)).EQ.2-I)11.15
IF (ABS(5D).GT.Y)12.30
YAMPINI(ABS(SD),Y+CY*ABS(ORC)) $ SC=SIGN(Y,SD) $ GC TC 30
I=1+3 $MU=PUL=PUL SCY=YACF*YAC $CCEF=CCEFU $CC TO 2C
RAUSCHINGER EFFECT PATH
C
10
15
        IF (1.EQ.4.AND.CRO.GT.C..CR.E.EC.6.ANC.CRC.LT.C.) 19.20
        1=10-1
        KKK-KKK 4 1
        JL (KKX !=19
20
        EMU=#U+1DY+RU-MU]+(A851.5+.5+SIGN(1.,CRC)+SDC/Y)}++xP
        KKK=KKK+1
        JI (KKK) = 20
        SD=SCO+COEF+EMU/MU
        SDAV=AHAX11-Y, (SC+SCG)/2.1+FLOAT(3-1/2)+AHIA1(Y, 150+SDO)/2.1+FLOAT
       1(1/2-2)
       EMU=ML+(BY+RG-FU)+(ABS(.5+.5+SIGN(1.,CRO)+SCAV/Y))++XP
        SC=SCO+COEF+EMU/ML
        IF (ABS(SD).GE.Y)22,30
22
        Y=AMINI(ABS(SC), Y+(SC-SIGN(Y.SE))/(SD-SDC)+DY+ABS(DRC))
        KKK=KKK+1
        JE (KKK) = 22
        SC=SIGNEY, SDJ 8 I=T-MAXOEE, 41
30
       CALL SSHTCHI3, ISHTCHI
       KKK=KKK+1
        JL (KKK) = 30
       GO TO (40,50)ISMTCH
PRINT 80.N.J.IOR.1.5DC.5D.YOR.1.CGEF,EPL.GRC.SDAV
PRINT 81,(JL(K),X=1,KKK)
40
50
       RETURN
       PRINT 100.1.N.J
99
      1,3+ Y=,E9.3,4H TO ,E9.3,6H COEF=,E10.3,4H PU=,E9.3,5H DRC=,E9.2,6H
      2 SDAY*, ELQ.3)
FORMAT (17H PATH IN BAUSCHI 2315)
FORMAT(15H H IS BRCAG, H=,12,3H A=,13,3H J=,12)
100
```

2500+

SUBROUTINE BAUSCHI (complete)

SUBROUTINE PUFPLOT(NIEXP, IDENTP)

THIS SUBROUTINE DIRECTS THE PLOTTING PROCEDURE
COMMON /PLOT/ IPLOT, IFPLOT(6), XJEC(7)
COMMON /PLOT/ IPLOT, IFPLOT(6), 1DUM(14)

CALL SECOND (BEG)
NTEX=NEXP \$19ENT=IDENTP
A""M=10H L, SEAMAN \$ CHARGE=10H6736-2 \$ XTFN=6HX-3587

C' XEROX
CALL HEADER (ANAME, CHARGE, XTEN)
IF (IFPLOT(1), EQ.O) GC TO 1
CALL TPLOTS(NTEY)
PRINT 2G
PEAD (7) N, NJEDIT, (JECIT(1), I=1,6), (XJED(1), I=1,6)
N=MINO(N-2,1000)
IF (IFPLOT(2), EQ.O) GC TO 2
CALL HISTORY \$PRINT 21
2 IF (IFPLOT(5)+IFPLOT(4)+IFPLOT(5), EC.O) GC TO 5
CALL HUGONIO \$ PRINT 22
5 CALL SECOND (END) \$ DUD-END-BEG \$ PRINT 25, DUR
RETURN
20 FORMAT(14H IPLOTS CALLED)
21 FORMAT(15H HISTORY CALLED)
22 FORMAT(15H HUGONIO CALLED)
23 FORMAT(15H HUGONIO CALLED)
24 FORMAT(15H TIME TO COMPLETE PLOTTING IS , F10,3,8H SECONDS)

MAIN PLOTTING SUBROUTINE: PUFPLOT (complete)

```
SUBROUTINE HISTORY
       COMMON /PLOT/ IPLOT, IFPLOT(6), LINES, IDUM(3), ING(9), IFLAG
DIMENSIGN T(1000), SEQV(1000, 9)
EQUIVALENCE (S, SEQV)
SMAX=SMIN=0... $ IFS=0 $ JT=0 $ ORIG=0.
                                               $ JT=0 $ ORIG=0.
$ IFLAG=[AB:'IFLAG)
           IF (IFLAG.LY.O) IFS=1
           DD 2 I=1.N
           READ (7) T(1), (SIF(1,J),J: 1,5;,(S(1,J),J=1,6)
           DO 2 J=1,9
IF (ING(J).EQ.O) GO TO 2
SMAN-AMAX1(SMIN.SEQV(I,'))
SMIN-AMINI(SMIN.SEQV(I,'))
  2
           CONTINUE
           THAX=IIN) STMIN=TILL SMARKED SIBLANK=O
  C INDIVIDUAL HISTORIES PLOTTED
           DO 30 J=1,8 SJN=J=6
IF (INGIJ).EQ.Q) CQ TQ.3Q
IF(J.15.6) GO TO 8
           ENCODE (30,211.SLABEL)
ENCODE (30,210.TITLE) JA, (BENT
          ENCODE (30,213,TITLE)XJED(J),IDENT
ENCODE (30,214,SLABEL)JEDIT(J)
JY=JT+1 $ JAD=J-JT
PRINT 240,(TITLE(M),M-1,3),TMIN,TMAX,SMIN,SMAX
  9
           CALL LINSET(TMAX.TMIN, SMAX, SMIN, TLABEL, SLABEL, TITLE, LINES)
           IPEN=0
           IFLAOSISEQVII.JID.GT.1E-3+SMAX) IPEN=1
CALL LINPLTIT(ID.SEQVII.J).MARK.IPEN)
           CONTINUE
  10
           CALL PENEND
           CONTINUE
  30
            IF (IFLAG.LE.1) GO TO 90
C SEVERAL JEDITS PLOTTED ON ONE GRAPH
ENCODE (30,2(1,5LABEL)
ENCODE (30,215,TITLE) DENT
            PRINT 240, (TITLE(M), M=1, 3), TOIN, THAX, SHIN, SHAX
           ## (ING(J).EQ.O) GO TO 60

IF (J. CQ.IFLAG) GO TO 42

DO 35 ID=1.5

IF (ING(J+ID).NE.O) GO TO 42

CONTINUE $ PRINT 250,J
           CONTINUE
                                                    $ RETURN
35
42
           DG 55 I=M,N
IF(K.EQ.O.AND.S(I.J).LT.1E-3*SMAX)GD TO 55
                  # IF(J.EQ.IFLAG) GO TO 54
           IF (L.NE.M) GO TO 44
           IF (S(1,J*10).GT.1E-3*SMAX)L=1
           IF (IFS.EQ.1)48,46
            IF (S(1,J).LT.1.05+S(1,J+10)) GO TO 56
           IF (I.EQ.N) GO TO 56
CALL LIMPLT (T(I),S(I,J),MARK,IPEN)
CONTINUE $ GO TO 60
CALL LIMPLT (T(L),S(L,J+ID),MARK,IBLANK)
   48
54
   54
   60
           CONTINUE
           CALL PENEND
   90
            RETURN
           FORMAT (17HINTERFACE STRESS 11.8M. IDENT=14)
FORMAT(30H STRESS (KILOBARS)
   210
   211
           FORMAT(30H TIME (MICROSECONDS)
FORMAT (11MHTSTORY, X=F7-4,0H CP. 10=14)
FORMAT(15HSTRESS IN CELL ,13,12H (KILOBARS) )
FORMAT(26H HISTORIES AT JEDITS 10 =,14)
   213
  214
215
           FORMAT 17H PLOT -- , 3A10,4H T= , F8.4,4H TO , F8.4,4H S= , F9.4,4H TO ,
          1F9.4)
FORMAT (24H COMBINED PLOT ERROR, J=.11)
  250
            END
```

PLOTTING SUBROUTINE: HISTORY (complete)

```
SUBROUTINE HUGONIC

THIS REUTINE PRODUCES GRAPHS OF STRESS, RESULTANY STRESS, AND DEVIATORIC

STRESS VERSUS SPECIFIC VOLUME

COMMON 10ENT, N, N, DEDIT, JEDIT(6), X, JEDI6), S(1000,6), V(1000,3),

1SLABEL(3), VLABEL(3), TITLE(3), VMAX(3), VMIN(3), VMAX(3), VMAX(3),
                                                        COMMON /PLOT/ IPLOT , IFPLOT (6) . LINES , IDUM (13)
                                                        DIMENSION VP(1000,3)
                                                       MARK=0 $ E=1
ENCODE (30,210,VLAREL)
                                                        (1) CONTINUAL CO
                                                     C(L,1)qv,(L)xAMqv)1"AMA=(L)xAMqv2 (C,,1)v,(L)xAMv11xAMa=(L)XAMv
()L,1)qv,(L)xAMqv)1xAMa=(L)xAMqv2
()L,1)qv,(L)xAMqv)1xAMa=(L)xAMqv2
()L,1)qv,(L)xAMqv1xAMqv2
()L,1)qv,(L)xAMqv2
                                            IF(IFPLOT(3).EQ.O) GO TO 30
IF(IFPLOT(2).NE.O) GO TO 10
READ STRESS FROM TAPE
                                                   READ (7) TI.(S(1,J),J=1,6),F1,F2,F3
NP=MINO(NJEDIT, IFPLOT(L+2))
       10
                                                     SMAX=SMIN=O
                                                   DC 12 J=1,NP
DO 12 I=1,N
                                                     SMAX=AMAXI(SMAX+S(I+J)) & SMIN=AMIN1(SMIN+S(I+J))
                                                   CONTINUE
                                            PLOT HUSONIOTS
                                                  CO 20 J=1,NP
CO TO (21,22,23) L
ENCODE (30,211,SLABEL) JEDIT(J)
ENCODE (30,212,SLABEL) JEDIT(J)
ENCODE (30,213,SLABEL) JEDIT(J)
ENCODE (30,215,TITLE)XJED(J),IDENT
                                                                                                                                                                                                                                                                                                              $ GO TO 25
$ GO TO 25
                                                   IF(J.GT.3) GO TO LT
PRINT 240, (TITLE(M), M=1, 3), VMIN(J), VMAX(J), SMIN, SWAY
                                                CALL LINSET(VMAX(J), VMIN(J), SMAX, SMIN, VLABEL, SLAGEL, ITLE, LINES)
DO 15 1=1, N & IPEN-MINO(I-1, 1)
                                                  CALL LINPLT(V(1,J).S(1,J),MARK, IPEN) $GC TO 20
                                               PRINT 240. (TITLE(M), M=1,3), VPMIN(K), VPMAX(K), SMIN, SMA:
CALL LINSET(VPMAX(K), VFMIN(K), SMAX, SMIN, VLABEL, SLABEL, TITLE-LINES,
DO 18 1~1,N $ IPEN=MINO(I-1,1)
                                               DO 18 1=1.N $ IPEN=MINO(I-1.1)
CALL LINPLT(VP(I,K),S(I,J),MARK,IPEN)
CALL PENEND
 20
                                              GO TO (30,40,50) L

IF (IFPLOT(4),EQ.O) (" TO 40

READ RESULTANT STRESS FROM TAPE
  30
                                              L=2 & DO 32 I=1.N
READ (9) (S(I,J),J=1.6)
 32
                                            GO TO 10
IF(IFPLOT(5).EQ.0) GO TO 50
 40
                                   RFAD DEVIATORIC STRESS FROM TAPE
L=3 $ DO 42 :=1.N
                                              READ (11) (S(I,J),J=1,6)
                                         READ (11) (S(I,J),J=1,6)
CD TO 10
RETURN
FORMAT (30H SPECIFIC VOLUME (CM3/GRAM3) }
FORMAT (19HSTRESS IN CFLL ,I3,12M (KILOBARS) )
TORMAT (19HRESULTANT STRESS J=,I3,8H (KBARS))
FORMAT (19HSTRESS DEVIATOR, J=,I3,8H (KBARS))
FORMAT (8HHUG., X=F7.4,11H CM, IDENT=14)
FORMAT (8HUG., X=F7.4,11H CM, IDENT=14)
FORMAT (7H PLOT=,3A10,4H V=,F8,5,4H IC ,F8.5,4H S=,F9.4,4H TO ,IF9.4)
 50
211
 212
215
                                     1F9.4)
                                                  NΩ
```

PLOTTING SUBROUTINE: HUGONIO (complete)

```
CUBRCUTINE TRUDISINTEX)

C. THIS BOUTINE PRODUCES STRESS VERSUS DISTANCE GRAPHS FOR DESIRED TEDITS
CUMMON IDENT, X (300, 15:. S (300, 15), X LABEL (3), Y LABEL (3), TITLE (3),
                  INPT(15).XJ8(6)
                    TOPING TO THE COMMON PLOTA PLOTAGE TO THE COMMON APLOTA PLOTAGE TO THE COMMON PLOTAGE TO THE COMMON TEXT THE C
 C INDIVIDUAL TEDITS PLOTTED
4CODE (30,211,XLABEL)
DG 100 1=1,NTEX
                     READ (7) STED NC TIME NPTS
                                                                                                                        SNPTS1=NPTS+L
                     10
                     READ (7) XIIP.IG1.SIIP.IG1.R
                    IF (IP.EQ.1) 11,12
XMIN=X(IP,IG) $ SMAX=SMIN=S(IP,IG)
  11
                     IFIIG.EC.1) XMIN1=XMIY
                                                                                                    $ GO TO 13
  12
                     XMAX=X(IP. IG)
                     SMAX=AMAX1(SMAX,S(IP, EG)) & SMIN=AMIN1(SMIN,S(IP, EG))
                     CONTINUE
  13
                    ENCODE (30.210.TITLE) IG, TIME, IDENT
ENCODE (30.212, SLABEL) NC
SMA=AMAXI(SMA, SMAX) & SMI=AMINI(SMI, SMIN)
PRINT 240. (TITLE(M), M=1, 3), XMIN, XMAX, SMI, SMA
                     CALL LINSET (XMAX,XMIN,SMA,SMI,XLABEL,SLABEL,T;TLE,LINES)
CALL LINPL: [X(1,IG),S(1,IG),MARK,IBLANK)
                     DO 15 IP=2.NPTS
                    CALL LINPLTIXB, SHI, PARK, I BLANK)
                    SINCR*(SMA-SMI)/75. $ 58*SMI

DO 70 IN=1.75 $ INK*MOD(IN.2)

CALL LINPLT (XB.SE, MARK, INK)

CALL PENEND $ GO FU 100

DO 91 IP=1,NPTS1
                                                                                                                                  $ $8#SB+SINCR
  80.
  90
                     READ 171
  100
                     CONTINUE
                  IFITELAG.LE.1) GO TO 150
SEVERAL TEDITS PLOTTED ON ONE GRAPH
IF (1G.LT.2) GO TO 145
                     IGPLCT=MING(IFLAG, IG)
                     IBEG# 1
                     EMCODE (30,220,117LE) 18EG, IGPLOT, IDENT
                     ENCODE (30,221,5LABEL)
PRINT 240,ITILE(M).M=1,3),XMIN,XMAX,SMI,SMA
GALL LINSET(XMAX,XM3NI,SMA,SMI,XLABEL,FLIRES)
GALL EINPLT(X(1,1),S(1,1),MARK,IBLANK)
                     LP=MP=1
                     00 130 L=18EG.16PLOT
                     MP=MPT(L)
                     DO 120 IP=MP.NP
                     IF (IFS.EQ.1) 120,115
                     ## (LP.NE.MP.OR.L.FC.IGIGO YO 120

## (SEEP,L).LY..9#SMA.AND.S(EP,L+1).G1.J(EP,L)#1.05) LP#EP
                    CALL LIMPLT (X(IP,L),S(IP,L),MARK,IPEN)
IF(L.EC.IG) GU TO 130
CALL LIMPLT (X(LP,L+1),S(LP,L+1),MARK,IBLANK)
 120
                     PP-LP
                     CONTENUE
  130
                    CALL PENEND
                     IF (IGPLOT.GE. IG-1) GO TO 150
IBEG-IGPLOT+1 $ IGPLOT-IGPLOT+MENO(IFLAG,IG-IGPLOT)
                    GO TO 105
                    PRINT 146
 150
                    RF TURN
                   FORMAT (23H ONLY ONE TEDIT GRAPHED)
FORMAT (5HTEDIT, 12, 1H, F7, 4, 11H MUSECS 1D=14)
FORMAT (30H POSITION (CENTIMETERS) )
FORMAT (15HSTRESS AT CYCLE, 14, 11H (KILOBARS))
FORMAT (4M TEDITS 12, 6H THRU 12, 8H, 1DENT=14)
FORMAT (30H STRESS (KILOBARS) )
  146
  210
 212
  220
 240
                    FORMAT LTH PLOT--, 3A10,4H X=, F8,4,4H TO , F8,4,4H 3*, F9,4,4H TO ,
                  1F9.4)
                    END
```

#### APPENDIX II

## QUASI-STATIC AND ACOUSTIC DATA

## 1. Introduction

As an adjunct to the dynamic measurements, a series of quasi-static acoustic measurements were made to assist in characterizing the metals studied. The purposes of these tests were to determine:

- the quasi-static moduli and yield strength,
- the variation of quasi-static properties with specimen thickness,
- the difference in quasi-static properties measured parallel and transverse to the direction of rolling,
- the effect of testing speed on the quasi-static properties, and
- the effect of the sign of the loading (tension or compression) on the properties.

For these purposes an extensive test program was laid out for the five alloys under study. The scope of the quasi-static experiments is partially indicated by the listing of specimens in Tables 13 and 14. Both tensile and compression tests were performed. Some compression tests were conducted on all three directions (parallel and transverse to rolling, and thickness) of the material. Tensile tests were conducted over a wide range of thicknesses and at three testing rates for one thickness.

## 2. Test Descriptions

## a. Tensile Tests

The tensile tests were conducted on the usual dog-bone-shaped test specimens instrumented with either strain gages or an extensometer. Two sets of tests were performed: one series at several thicknesses but a constant testing speed of 0.05 in./in./min. and a second series at a single thickness but three testing speeds (0.5, 0.05, and 0.005 in./in./min). The second series was instrumented with both longitudinal and transverse strain gages so that Poisson's ratio could be determined.

The tests were generally conducted with a continuous recording of load and deflection so that Young's modulus, the yield strength, ultimate strength, and breaking strength could be determined.

Table 13
ALUMINUM: NUMBERS OF SPECIMENS TESTED

Alloy	Thickness	С	ompressi	on	Tet	ısile
	(in.)	P.*	Tr †	Th <sup>‡</sup>	p*	Tx †
6061 <b>-</b> T6	1/16				2	2
	1/8	ļ		2	2	2
	3/16		Ì		2	2
	1/4	3	3	2	(9)	(9)
	3/8				2	2
	1/2	2	2	4	2	2
	3/4				1	1
	)-1/4	1	1	3	1	1
<del></del>		A James Area Company				
2024-T8	1/16		İ		2	2
	3/32	ļ			2	2
	1/8			2	2(9)	2(9)
	3/16			<u> </u>	2	2
	1/4	3	2	2		
	3/8	1			2	2
	1/2	2	2	3	2	2
	3/4				1	1
	1-1/4	1	1	4	1	1
		]				

Parenthetical numbers refer to tests conducted at three testing speeds. Other tests were conducted at a single speed.

<sup>\*</sup> Parallel to rolling.

<sup>†</sup> Transverse to rolling.

<sup>&</sup>lt;sup>‡</sup>Thickness direction.

Table 14
TITANIUM: NUMBERS OF SPECIMENS TESTED

Alloy	Thickness		Compressio	n	To	nsile
	(in.)	P *	Tr 1	Th #	.р.*	.r †
Type 50A	1/16				2	1
	1/8		.,	1	(3)	(3)
	3/16	1	1		1	1
	3/8	1	1	2	1	1
	3/4	1	1	.3	1	1
	1	1	1	4	1	1
T1-6A1-4V	1/16				1	1
	1/8			4	(3)	(3)
	1/4		1	4	1	1
	3/4	1	1	.3	1	1
	1-1/8	1	1	4 ~	1	1
Ti-13Cr-11V-3A1	1/16	;		# 1 ·	1	1
	1/8			4	(3)	(3)
	1/4	1	1	:3 - :-	1	1
	3/4	1	1	3	1	1
	1-1/8	ı	1	4	1	1
		,	:			;

Parenthetical numbers refer to tests conducted at three testing speeds. Other tests were conducted at a single speed.

<sup>\*</sup> Parallel to rolling.

 $<sup>^{\</sup>dagger}$ Transverse to rolling.

<sup>&</sup>lt;sup>‡</sup>Thickness direction.

# b. Compression Tests

The compression tests were conducted on dog-bone-shaped specimens similar to those for tension tests and at the rate of 0.05 in./in./min. The thick portions at the ends of the specimens were provided to give lateral support to prevent buckling. The specimens were instrumented either with an extensometer or a pair of longitudinal strain gages.

The tests were conducted until buckling occurred or until the capacity of the machine was reached. It is believed that all the exhibited yield data represent yielding of the material and not premature buckling.

## c. Thickness Compression

Specimens about 1 inch thick were manufactured for determining properties in the thickness direction. The specimens were formed by stacking together a number of 1 inch square coupous from each sheet of material. Strain was measured with an extensometer recording the overall compression of the specimen. These specimens were loaded at 0.05 in./in./min.

The load-deflection curves show an initial soft portion as the coupons comprising the specimen are being brought into contact, then a straight elastic region, and finally yielding. Post-yield behavior was not interpreted for these tests.

### d. Acoustic Tests

Dilatational and shear velocities were measured on 1 inch square coupons from each of the materials. The tests were performed for the thickness direction on all specimen thicknesses, and in all three directions on specimens over 1 inch thick.

## 3. Regults of Quasi-Static Tests

A total of 234 static tests was conducted on specimens of the three titanium and two aluminum alloys to characterize the materials. Average magnitudes of the clastic properties determined from the test program are listed in Tables 15 and 16, and the results of the parameter studies are given in Tables 17 and 18. For the titanium there was considerable variability mong results from samples of different thicknesses; the

Table 15
TITANIUM DATA FROM STATIC TESTS (AVERAGE VALUES)

Alloy	Young's Modulus* kbar (psi x 10 <sup>3</sup> )	Plastic Modulus <sup>†</sup> kbar (psi x 10 <sup>3</sup> )	Yield Strength <sup>‡</sup> kbar (psi x 10 <sup>2</sup> )	Ultimate Strength <sup>§</sup> kbar (psi x 10 <sup>3</sup> )	Poisson's Ratio
Туре 50А	1070 (15,500)	66 (960)	3.5 (50)	4.1 (65)	0.26
T1-6A1- IV	1210 (17,500)	7 (100)	9.5 (138)	9.6 (139)	0.29
Ti-13Cr-11V-3A	1 1040 (15,100)	6 (80)	9.3 (135)	9.4 (136)	0.27

<sup>\*</sup>Coefficient of variation (standard deviation/mean value) about 2 percent for samples of a single thickness.

<sup>†</sup>Slope of line from yield point to lpcrcent strain. Highly variable; negative values were obtained from alloy Ti-13Cr-11V-3Al in many tests.

<sup>\*</sup>An offset of 0.2 percent.

<sup>§</sup> Obtained from peak of stress-strain curve before decrease to breaking.

Table 16
ALUMINUM DATA FROM STATIC TESTS (AVERAGE VALUES)

Alloy	Young s Modulus * kbar (psi x 10 <sup>3</sup> )	Plastic Modulus kbar (psi x 10 <sup>3</sup> )	Yield Strength <sup>‡</sup> kbar (psi x 10 <sup>3</sup> )	Ultimate § Strength kbar (psi x 10 <sup>3</sup> )	Poisson's Ratio
6061 <b>-</b> T6	710 (10, 300)	103 (1490)	2.8 (40.7)	3.1 (45.1)	0.32
2024-T8	745 (10,800)	32 (460)	4.4 (63.6)	4.9 (71.0)	0.33

<sup>\*</sup> Coefficient of variation about 1 percent for specimens of a single thickness.

<sup>†</sup> Average slope of stress-strain curve from yield to 1 percent strain.

<sup>\* 0.2</sup> percent offset; coefficient of variation up to 1.6 percent for one thickness and direction of rolling, up to 6 percent for all thicknesses.

Coefficient of variation of  $^{\circ}$  7 percent or less for one thickness and direction of rolling, 3 percent or less for all thicknesses.

Table 17
STATIC TEST PARAMETER VARIATION FOR TITANIUM

<u> </u>	Parameter Varied	Young's Modulus, E	Yield Strength, * Y	Ultimate Strength, U
L	Thickness	Maries of different thick- nesses with a trend toward decreasing modulus with increasing thicknesses	Marked variation among samples of different thick-nesses, but no apparent trend attributable to thickness	Marked variation among samples of different thicknesses, but no apparent trend attributable to thickness
	Parallel vs transverse to direction of rolling	No significant effect	50A: 20%-30% higher for parallel 6-4: slightly higher for transverse 13-11-3: no significant difference	No significant difference
······································	Tension vs compression	No significant effect	2%-3% higher for compression	(Nc ultimate strength observed in compression)
	Strain rates: 0.005, 0.05, or 0.5 in./in./ min	No effect	50A:13% increase from lowest to highest rate 6-4: 6.5% increase 13-11-3: 4% increase	50A: 9% increase from lowest to highest rate 6-4: 6% increase 13-11-3: 11% increase

\* Offset of 0.2 percent.

Table 18 STATIC TEST PARAMETER VARIATION FOR ALUMINUM

1113.4

Parameter Varied	Young's Modulus E	Yield Strength, * Y	Ultimate Strength, U
Thickness	No effect	6061: No effect 2024: 10% increase hrough range of thicknesses	6061: No effect 2024: 10% increase through range of thicknesses
Parallel vs transverse to direction of rolling	No effect	6061: 4% higher parallel 2024: No effect	No effect
Tension vs compression	No significant effect	6061: 4% higher in tension 2024: No effect	(No ultimate strength observed in compression)
Strain rates: 0.005, 0.05, or 0.5 in./in./min.	No effect	No effect	6061: 2% increase with strain rate 2024: No effect

\* Offset of 0.2 percent.

coefficients of variation (standard deviation/mean value) of the yield strength Y and ultimate strength U were usually about 5 percent and in some cases 10 percent. This difference is attributed to real variations among specimens, despite the special care taken in acquiring material of controlled quality. The aluminum exhibited a coefficient of variation of 1 to 2 percent for tests at a single thickness and 3 to 6 percent for tests at all thicknesses.

# 4. Elastic Constants from Acoustic Velocity Measurements

Acoustic velocity measurements were made with samples of the aluminum and titanium materials to determine the elastic constants of the materials at nominally zero stress levels. The tests were conducted by transmitting longitudinal (compressional) and shear pulses through thin specimens and measuring the transit times. These transit times were then used to compute Young's modulus E and Poisson's ratio  $\nu$  for each material.

ones for linearly elastic, homogeneous, isotropic materials. The samples did not qualify as isotropic because their properties were different in the three principal directions: parallel to rolling, transverse to rolling, and in the thickness direction. These differences were small however, and were neglected in the analysis. This neglect is testament to the assumption that wave propagation in one direction is but slightly affected by anisotropy in the other two directions. With the assumption of isotropy we can compute longitudinal and shear acoustic velocities  $C_L$  and  $C_S$  from the travel times of the waves through the samples. These sound velocities are related to the elastic constants as follows:

$$C_L = \sqrt{\frac{E}{\rho} \frac{(1-\nu)}{(1-2\nu)(1+\nu)}}$$
 (106)

$$C_{S} = \sqrt{\frac{E}{\rho} \frac{1}{2(1+\nu)}}$$
 (107)

where ho is the material density. Equations (106) and (107) can be solved for E and u , obtaining

$$\nu = \frac{(c_L/c_S)^2 - 2}{2(c_L/c_S)^3 - 2}$$
 (108)

$$E = 2(1 + \nu) C_S^2 \rho$$
 (109)

$$E = \frac{(1-2\nu)(1+\nu)}{(1-\nu)} c_L^2 \rho$$
 (110)

The measured acoustic velocities and the elastic constants derived therefrom are exhibited in Table 19 for the five alloys tested.

The elastic constants calculated from the sound velocities are not directly proportional to these velocities but are related in a complex manner to the square of the velocities. Because of this complicated relationship, it is not obvious how the uncertainties in the measured velocities contribute to uncertainties in the elastic constants. The following analysis represents an attempt to define the uncertainties in those constants, based on estimated uncertainties in the measured velocities.

The probability distributions for the longitudinal and shear wave velocities were assumed to be normal for convenience in the analysis and for lack of a better assumption. The probability density functions for all the variables are certainly continuous functions. For simplicity in the analysis, however, we approximated distributions of the velocities by a small number of points with equal probability. The points were selected so that the means and standard deviations of the discrete and continuous distributions were equal. Selected points were obtained by dividing the continuous distribution into 10 intervals of equal probability, that is, with equal areas under the probability distribution curve. With this formulation we obtained 10 discrete, equally probable values for each of the two wave velocities. We can combine these to form 100 equally probable values for each of the elastic constants E and  $\nu$ . The resulting values of E and  $\nu$  form discrete approximations to

Table 19
ACOUSTIC VELOCITY MEASUREMENTS

	¥a	Dilational (mm/µsec	Velocity )		-	Shear Velocity (mm/µsec)	ocity c)		Elastic Constants	ic nts	Density
Alloy	thick.	Direction thick. longit.	trans.	Avg.	D thick.	Direction thick.   longit.	trans.	Àvg.	۵	E (kbar)	E (Kbar) (g cm <sup>2</sup> )
Aluminum				1							
6061-T6	6.41	6.48	6.48	6.44	3.18	3.04	3.17	3.15	0.337	723	2.709
2024-T8	6.43	6.38	6.38	6.41	3.17	3.14	3.12	3.16	0.339	745	2.785
											Atti
Titanium											
Type 50.4	6.13	6.12	6.02	6.13	3.13	3.00	2.99	3.10	0.328	1175	4.507
T1-6A1-4V	6.28	6.14	6.26	6.23	3.27	3.26	3.23	3.24	0.314	1240	4.424
T1-13Cr-11V-3A1	5.62	5.98	5.97	5.72	2.80	2.79	7.80	2.80	0.332	1025	4.839

the desired continuous distributions. Calculations were made using nominal values for  $\mathbf{C}_L$  and  $\mathbf{C}_S$ 

The coefficients of variation (standard deviation divided by mean value) of  $C_L$  and  $C_S$  were assumed to be 1 and 2 percent respectively. The resulting coefficients of variation for E and  $\nu$  are given in Table 20. The mean values of both the elastic constants coincide with those computed from the mean values of  $C_L$  and  $C_S$ . The coefficients of variation are somewhat larger than those for either  $C_L$  or  $C_S$ . In general, we can probably obtain E and  $\nu$  with somewhat less precision than we can obtain either of the sound velocities.

Table 20

ILLUSTRATION OF VARIABILITY OF ELASTIC CONSTANTS

DERIVED FROM ACOUSTIC MEASUREMENTS

Measurement*	Mean	Coefficient of Variation (%)
C <sub>L</sub> (mm/µsec) <sup>‡</sup>	6.0	1
C <sub>L</sub> (mm/µsec) <sup>‡</sup> C <sub>S</sub> (mm/µsec) <sup>‡</sup> E (kbar) <sup>§</sup>	3.0	2
E (kbar) §	720	3.14
v §	0.333	2.81

<sup>\*</sup>  $C_L$  and  $C_S$  are arbitraly; density was taken as  $3 \text{ g/cm}^3$ .

<sup>†</sup> Coefficient of variation = standard deviation/mean.

<sup>\*</sup> Assumed.

Computed.

# 5. Titanium Properties Tests by the Manufacturer

The manufacturer, Titanium Metals Corporation of America, supplied a list of test data for each thickness of the titanium alloys. Chemical analyses showed about 0.11 percent 0 for all three types and 0.02 percent N in Ti-13Cr-11V-3Al titanium and 0.01 percent N in the other two types. The tensile test values supplied are shown in Tables 21, 22, and 23. For the Type 50A (pure alpha) titanium, three samples of each thickness were tested and all values reported in Table 19. The variation of yield among the three samples at each thickness was studied. The coefficient of variation (standard deviation/mean value) at each thickness varied from 5 to 12%. There was a clear trend toward decreasing yield strength with increasing thickness; average yield values varied from 54,000 psi down to 47,000 psi.

Only average strengths from several tests were reported for Ti-6Al-4V and Ti-13Cr-11V-3Al titanium. However, these were differentiated by direction of rolling. For the Ti-6Al-4V alloy, the yield strength was constantly 3000 to 4000 psi higher when tested transverse to the direction of rolling than for parallel to rolling. There was a clear trend of decreasing strength with increasing thickness. The Ti-13Cr-11V-34l alloy was also stronger in the transverse direction than parallel to the rolling direction. The yield data of this alloy varied considerably from sheet to sheet so that no trend of strength variation with thickness could be established.

Table 21
MANUFACTURER'S DATA FOR TYPE 50A TITANIUM

Thickness (in.)	Yield Strength (psi x 103)	Ultimate Strength (psi x 10 <sup>3</sup> )	Elongation (%)
0.0625	56.0, 40.8, 57.3	70.6, 68.7, 71.6	28.0, 26.5, 28.0
0.125	49.9, 48.1, 58.1	66.4, 65.8, 66.8	27.0, 27.0, 29.0
0.1875	47.6, 60.7	72.8, 72.5	26.5, 27.0
0.375	46.5, 44.3, 50.3	68.5, 67.2, 70.2	32.5, 31.0, 34.0
0.750	44.5, 40.8, 49.1	65.0, 63.8, 66.4	36.0, 35.0, 37.0
1.0	46.5, 42.9, 52.3	68.0, 65.5, 69.6	54.5, 53.0, 56.5
Average	49.7	68.0	34.6

Table 22

MANUFACTURER'S DATA FOR T1-6A1-4V TITANIUM

Thickness (in.)	Parallel	Strength  Transverse* x 10 <sup>3</sup> )	Parallel	e Strength  Transverse* x 10 <sup>3</sup> )		ation   Transverse <sup>*</sup> %)
0.0625	134.7	137.7	141.5	146.0	14.0	12.0
0.125	132.5	144.7	136.6	145.7	15.0	15.0
0.250	133.7	138.0	139.1	142.6	16.0	14.5
0.750	126.8	130.9	134.3	137.4	15.0	15.5
1.125	121.9	126.1	129.3	131.9	13.5	16.5
Average	129.9	135.5	136.2	140.7	14.7	14.7

<sup>\*</sup>Parallel and transverse refer to the direction of rolling.

Table 23

MANUFACTURER'S DATA FOR Ti-13Cr-11V-3A1 TITANIUM

Parallel	Transverse *	Parallel	Transverse *	Parallel	(stion   Transverse *
126.5	135.7	131.7	138.8	20.0	14.5
129.8	132.2	135.2	137.7	21.0	17.5
130.6	137.8	141.1	146.R	24.0	12.0
126.8	132.0	131.8	139.6	24.0	16.5
125.3	129.5	132.3	136.6	18.5	13.5
127.8	133.4	134.4	139.9	21.5	. 14.8
The second of th	Parallel (psi 126.5 129.8 130.6 126.8 125.3	129.8     132.2       130.6     137.8       126.8     132.0       125.3     129.5	Parallel (psi x 103)     Parallel (psi x 103)       126.5     135.7     131.7       129.8     132.2     135.2       130.6     137.8     141.1       126.8     132.0     131.8       125.3     129.5     132.3	Parallel   Transverse* (psi x 103)       126.5     135.7     131.7     138.8       129.8     132.2     135.2     137.7       130.6     137.8     141.1     146.8       126.8     132.0     131.8     139.6       125.3     129.5     132.3     136.6	Parallel   Transverse * (psi x 103)     Parallel   Transverse * (psi x 103)       126.5     135.7     131.7     138.8     20.0       129.8     132.2     135.2     137.7     21.0       130.6     137.8     141.1     146.8     24.0       126.8     132.0     131.8     139.6     24.0       125.3     129.5     132.3     136.6     18.5

Parallel and transverse refer to the direction of rolling.

# 6. Discussion of Quasi-Static and Acoustic Results

The elastic properties of aluminum developed from the quasi-static tests agree with those from the acoustic tests to within 1 or 2 percent. A similar comparison for the titanium alloys shows that the values of Young's modulus agree within 3 percent for the Ti-6Al-4V and Ti-13Cr-11V-3Al alloys but that the acoustic modulus is 10 percent high for type 50A. This difference may indicate that the modulus decreases gradually with loading instead of maintaining a constant modulus up to the yield point. The values of Poisson's ratio derived from the acoustic tests are up to 20 percent higher than those derived from the quasi-static tests: this disagreement has not been explained.

A comparison of the manufacturer's test results and our data shows that there is good agreement for yield and ultimate strength for type 50A titanium. Our values for yield strength are about 3 percent higher for the Ti-6Al-4V and Ti-13Cr-11V-3Al alloys but the two sets of values for ultimate strength agree very well.

Our yield strength data on Ti-6Al-4V are about 25 percent below those of Maiden and Green (Ref. 8) for the same range of strain rate. This difference suggests that there were important differences in the retals tested by the two organizations, although both sets had the same normal composition and were fully annealed. The increase in the yield strength was about 3 percent per decade of strain rate according to the results of both organizations.

Our results for 6061-TE agree with those of Maiden and Green (Ref.8) in exhibiting a yield strength of 41,000 psi and no strain rate likect.

The quasi-static measurements have shown that the aluminum alloys are not strain-rate-dependent in the range of testing speeds used. The titanium alloys show a rate-dependence of strength: This rate effect could lead to a variation of the Hugoniot elastic limit in shock experiments. The noted sample-to-sample variability in properties strongly suggests that any test series should be conducted on material of a single sheet.

AFWL-TR-69-96

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Security Classification

	DOCUMENT CONT				
(Security classification of	of title, body of abstract and indexing	annotation must be e			
1. ORIGINATING ACTIVITY (Comp.	• -		2a. REPORT SE	CURITY CLASSIFICATION	
Stanford Research Ins	stitute		UNC	LASSIFIED	
Menlo Park, Californi	la 94025		26. GROUP		
3. REPORT TITLE			<u> </u>		
CLASSIFICATION OF MAT	CERIALS BY SHOCK PROPER	TIES			
4. DESCRIPTIVE NOTES (Type of	report and inclusive dates)			<del></del>	
May 1967-March 1969					
5. AUTHOR(3) (First name, middle	initial, lest name)		<del></del>		
	iams; J. T. Rosenberg;			inde	
6. REPORT DATE		78. TOTAL NO. OI	PAGES	76. NO. OF REFS	
November 1969		298		51	
SE. CONTRACT OR GRANT NO.	F29601-67-C-0087	94. ORIGINATOR'S	REPORT NUME	BER(5)	
b. PROJECT NO.	5719	AFWL-TR-69-96			
<sup>c.</sup> Subtask No.	RAS 1114	9b. OTHER REPOR	RT NO(S) (Any of	ther numbers that may be easigned	
d.		Contractor's	No.: SR	I Project PGU 6736	
10. DISTRIBUTION STATEMENTT	his document is subject	t to special	export co	ntrols and each trans-	
mittal to foreign gov	vernments or foreign na	tionals may	be made on	ly with prior approval	
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11. SUPPLEMENTARY NOTES	Topolor ,	12. SPONSORING N	ILITARY ACTI	VITY	
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3. ABSTRACT

(Distribution Limitation Statement No. 2)

A new experimental technique was developed for determining loading and unloading stress! volume paths directly from gage measurements; theoretical models were formulated for stress relaxation and the Bauschinger effect: Hugoniot information was generated from impact experiments on aluminum alloys, titanium alloys, and a woven quartz-phenolic; and the new experimental technique and Bauschinger calculations were applied to the aluminum alloys. The new experimental technique provides for measurement of complete loading and unloading paths rather than the discrete Hugoniot or release points previously obtained. The technique is applicable to the examination of nonsteady-state, nonisentropic flow, yield point phenomena, strain-hardening, the Bauschinger effect. and strain-rate (or stress-relaxation) effects. The technique is based on the entire stress or particle-velocity records obtained from a series of gages embedded in a specimen. Four stress relaxation models were implemented in the SRI PUFF wave propagation code, and computations were made to obtain representative results. The Bauschinger model implemented in the SRI PUFF code exhibits the smooth unloading adiabat and high rarefaction velocity observed in our experiments on 6061-T6 and 2024-T8 aluminum alloys Experiments with Ti-50A, Ti-6A1-4V, and Ti-13Cr-11V-3A1 titanium served to map Hugoniots from 15 to 750 kbar, indicate Hugoniot elastic limits, and show an alpha comega phase transformation at 50 kbar. Preliminary experiments on a three-dimensional woven quartz phenolic resulted in Hugoniot data from 10 to 200 kbar and indicated a shock-wave structure that is very different from that observed in homogeneous solids.

DD FO.M. 1473

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