

AD 680052

HIGH TEMPERATURE THERMAL EXPANSION OF UO₂ AND ThO₂

M. HOCH AND A. C. MOMIN

University of Cincinnati

TECHNICAL REPORT AFML-TR-68-219

OCTOBER 1968



えう

This document has been approved for public release and sale; its distribution is unlimited.

AIR FORCE MATERIALS LABORATORY AIR FORCE SYSTEMS COMMAND WRIGHT-PATTERSON AIR FORCE BASE, OHIO

NOTICE

When Government drawings, specifications, or other data are used for any purpose other than in connection with a definitely related Government procurement operation, the United States Government thereby incurs no responsibility nor any obligation whatsoever; and the fact that the Government may have formulated, furnished, or in any way supplied the said drawings, specifications, or other data, is not to be regarded by implication or otherwise as in any manner licensing the holder or any other person or corporation, or conveying any rights or permission to manufacture, use, or sell any patented invention that may in any way be related thereto.

This document has been approved for public release and sale; its distribution is unlimited.



Copies of this report should not be returned unless return is required by security considerations, contractual obligations, or notice on a specific document.

400 - November 1968 - CO455 - 58-1339

HIGH TEMPERATURE THERMAL EXPANSION OF UO₂ AND ThO₂

M. HOCH AND A. C. MOMIN

This document has been approved for public release and sale; its distribution is unlimited.

FOREWORD

This report was prepared by the University of Cincinnati under Contract No. F33615-67-C-1565. This contract was initiated under Project No. 7360, Task No. 736005.

The work was administered under the direction of the Air Force Materials Laboratory, Air Force Systems Command with Mr. Freeman F. Bentley as project engineer.

This report summarizes work performed from July 1967 through July 1968. Manuscript was released in July by the author for publication as an RTD technical report.

The work was performed at the University of Cincinnati with Dr. Michael Hoch serving as the principal investigator.

This report has been reviewed and is approved.

Freeman & Bentlup

Freeman F. Bentley () Chief, Analytical Branch Materials Physics Division Air Force Materials Laboratory

ii



ABSTRACT

The thermal expansion of uranium dioxide and thorium dioxide has been measured between 20 and 2100°C using high temperature x-ray diffraction techniques. The thermal expansion of UO_2 and ThO_2 as measured by x-ray diffraction is identical to that obtained by bulk expansion measurements. Because of this, and because the specific heat of UO_2 shows a rapid increase above 1700°C indicating a disorder, it must be concluded that the major structural defect is a Frenkel type disorder. This probably involves the oxygen atom moving from the tetrahedral into an empty octahedral position.

TABLE OF CONTENTS

																						Page
INTROD	UCI	TION	•	•	•	•	•	•	•	•	•	•	•	•	•	•	•	٠	•	•	•	1
EXPERI	MEN	TAL	PI	RO	CEI	บบเ	RE	•	•	•	•	•	•	٠	•	•	•	•	•	•	•	3
EXPERI	MEN	TAL	RI	S	UL	rs	Al	D	D	(SC	CUS	SSI	101	1	•	•	•	٠	•	•	•	5
А	•	UO2	٠	•	•	•	•	•	•	•	•	•	•	·	•	•	•	•	•	•	•	5
B		ThO	2	•	•	•	•	•	•	•	•	•	•	•	•	•	•	•	•	٠	•	8
refere	NCE	s.	•	•	•	•	•	٠	•	•	•	•	•	•	•	•	•	•	•	•	•	16

iv

INTRODUCTION

The importance of UO_2 and ThO_2 as fuel and fertile material in nuclear power reactors is well known. However, limited data exist in the literature on the high temperature physical properties of these materials. Thermal expansion data¹⁻⁷ on UO₂ have been obtained mostly by bulk expansion measurements using dialatometric, interferometric, or telemicroscopic techniques up to the melting point of UO2, and by x-ray diffraction techniques $^{8-12}$ up to 900°C. Recently, Baldock, Spindler, and Baker¹³ have obtained thermal expansion data using x-ray diffraction up to 2250°C. Their data are in general agreement with the bulk expansion measurements up to 1400°C; however, above this temperature Baldock, et al. 13 found that the x-ray thermal expansion was significantly lower than the bulk expansion. The reasons for this difference at higher temperatures were not clear and certain. Baldock, et al.¹³ suggest a large contribution by Schottky defects. The present work was therefore undertaken to elucidate this point. For this purpose, thermal expansion measurements using high temperature x-ray diffraction techniques were carried out on UO2 in vacuum over the temperature range 850-2100°C.

In order to check the results obtained with UO_2 , thermal expansion measurements were carried out on ThO₂ using high

temperature x-ray diffraction techniques, and the results compared with those of other investigators. $^{14-16}$

EXPERIMENTAL PROCEDURE

Equipment

The high temperature induction heated x-ray diffraction camera used in this work was the same as used by Hoch, Dean, Hwu, and Wolosin¹⁷ and by Wyder and Hoch.¹⁸ Temperature was measured with an L&N disappearing filament optical pyrometer.

Copper K_{α} radiation was used for obtaining high temperature x-ray diffraction patterns. Room temperature x-ray diffraction patterns were taken on a Norelco x-ray diffraction unit using CuK_a radiation.

An induction heated vacuum furnace was used to heat large quantities (5-10 grams) of UO_2 to elevated temperatures in a tungsten crucible to study the variation of stoichiometry with heating temperature.

Materials

The uranium dioxide powder of 99.9% purity was supplied by K & K Laboratories, Plainview, New York (Lot 31833); its average particle size was between 100 and 150 mesh. Thorium dioxide, 99.9% ThO₂, was obtained from Fairmount Chemical Company, Inc., Newark, New Jersey.

Temperature Calibration

To eliminate the error due to the emmissivity of the sample, the temperature of a black-body hole put in the place

of the x-ray diffraction sample was measured. For this purpose, a hole 0.047" in diameter by 0.063" deep was drilled into a sample of 0.15" diameter and 0.16" length. Thus the correction was evaluated under conditions identical to those when x-ray diffraction patterns were taken. For the absorption correction of the glass window, the curve developed in this laboratory was used.

Sample Preparation and Operation

The samples were pressed without any binder and were compacted into cylindrical rods of 1/16" diameter and about 3/16" length. The sample was then placed into the high temperature x-ray diffraction camera. The camera was evacuated with a mechanical fore pump and an oil diffusion pump to a vacuum of $2x10^{-5}$ torr. After heating to the elevated temperature, the sample was kept at this temperature for 1/2 hour before the 5-hour long exposure was taken. For each run a fresh sample was used.

Measurement of Lattice Parameters

The lattice parameters were calculated by employing the Nelson-Riley extrapolation technique.¹⁹ The accuracy of the lattice parameter measurements in the high temperature x-ray diffraction camera is ± 0.003 Å; that in the Norelco room temperature camera is ± 0.001 Å.

EXPERIMENTAL RESULTS AND DISCUSSION

A. UO2

The oxygen-to-uranium ratios of several UO_2 samples were determined by the oxidation method and are given in Table I. Weighed amounts of the sample were heated in air at 650°C to constant weight, and the ratios then calculated from the weight gain due to oxidation to U_3O_8 . Samples 1, 2, 3, and 4 were treated in this manner; the average oxygen-to-uranium ratio of these samples was 2.067±0.002.

According to the calculation of Hoch and Furman,²⁰ the oxygen partial pressure above $UO_{2.07}$ is quite high $(3.73 \times 10^{-7}$ atm at 1244°C) and thus heating the sample should cause $UO_{2.07}$ to decompose and approach the stoichiometric composition $UO_{2.0}$ as it is heated to higher and higher temperatures. To check this, samples of $UO_{2.07}$ were heated in a vacuum of 10^{-6} torr for 5 hours to different temperatures and cooled rapidly in vacuum. Both in this measurement and in the high temperature x-ray diffraction camera, "cooling rapidly" means shutting off the induction heating power and letting the sample cool by radiation in vacuum. As the mass of the heated parts is very small, the rate of cooling is fairly rapid. The samples thus obtained were analyzed for oxygento-uranium ratio and the results are given as Samples 5, 6, and 7 in Table I. As can be expected after heating to high

temperatures, the oxygen-to-uranium ratio becomes lower and the sample heated to 1960°C has the stoichiometric composition.

The lattice parameter measurements on UO_2 samples over the temperature range of 850 to 2000°C are summarized in Table II. The room temperature lattice parameters of the sample as received and unheated (Run No. 1) and of the samples after heating to 925, 1244, 1542, and 1960°C in vacuum and cooling to room temperature in vacuum (Runs 6a, 8a, 13a, and 23a) are identical. This is expected because the difference in lattice parameter between $UO_{2.00}$ and $UO_{2.07}$ at room temperature is only 0.005Å.²³

To show that the small change in stoichiometry has a negligible effect on the lattice determination of the thermal expansion coefficient, x-ray diffraction patterns were taken on two samples at 845°C and 927°C after the samples had been heated for 5 hours to 1960°C and thus had the stoichiometric composition $UO_{2.0}$. These two points are points 24 and 25 in Table II and are plotted with a different sign in Figure 1. They do not differ from the other data points.

The data in Table II were converted into percent linear expansion and plotted together with those of Conway, et al.⁷ and Baldock, et al.¹³ in Figure 1. As can be seen in Figure 1, the present data obtained by x-ray diffraction techniques

and those of Conway^7 obtained by bulk expansion measurements agree well over the whole experimental range, but the present data disagree with that of Baldock.¹³ As the specific heat of UO₂ shows a rapid increase above 1700°C,²¹ indicating some kind of lattice disorder, it must be concluded that the lattice disorder in question is of the Frenkel type. For a Frenkel type disorder the thermal expansion determined by x-ray diffraction and by bulk measurements should be equal. The Frenkel disorder which suggests itself immediately is that of an oxygen atom in a tetrahedral position moving into an empty octahedral site. This type of disorder should be present, as it is relatively easy to prepare hyper- and hypo-stoichiometric UO₂.

It is impossible to understand and explain the difference in thermal expansion measurements obtained by x-ray diffraction between Baldock, et al.¹³ and this research. Baldock¹³ only used two specimens whereas in this work a new sample was used for every lattice parameter measurement. The precision in Baldock's determination is greater than in the present data; however, the scatter in the thermal expansion curve in Figure 1 of Baldock's data is not smaller than the scatter of the present results. The possibility exists that the thermocouple used by Baldock¹³ (W/W-26Re thermocouple) may have deteriorated during the measurements.

An error of 300°C at 2200°C in Baldock's temperature measurements would be required to bring his highest data point onto the curve of Conway.⁷

Thus, the thermal expansion of $UO_{2.0}$ is best represented by the equation given by Conway, et al.⁷

* Expansion = $6.797 \times 10^{-7} \text{T} + 2.896 \times 10^{-7} \text{T}^2 - 1.723 \times 10^{-2}$, T in °C.

B. ThO₂

The lattice parameter measurements of ThO_2 are summarized in Table III and plotted in Figure 2 in terms of the linear thermal expansion as a function of temperature. This figure also contains the x-ray thermal expansion measurements of Aronson, et al.¹⁴ and the bulk expansion measurements of Geller and Yavorsky¹⁵ and Ohnysty and Rose.¹⁶ The agreement on thermal expansion of ThO_2 among the four authors is extremely good. From the data of Ohnysty and Rose¹⁶ the following equation is derived for the thermal expansion of ThO_2 :

* Expansion = $8.383 \times 10^{-4} \text{T} + 0.9995 \times 10^{-7} \text{T} - 2.106 \times 10^{-2}$, T in °C.

The bulk and x-ray expansion data on ThO₂ agree with each other. The only high temperature heat content data

available²² do not show a rapid rise. Thus, the disorder in ThO_2 is not yet defined.

To compare the thermal expansions of ThO_2 and UO_2 at elevated temperatures, the ratio of their thermal expansion was plotted in Figure 3. As can be seen, above 700°C where the measurements of Conway, et al.⁷ and ours were carried out, the ratio is a smooth curve somewhat concave downward. From this it can be concluded that the thermal expansion of UO_2 is quite similar to that of ThO_2 .

TABLE I

•

,

Oxygen-to-Uranium Ratio of Uranium Dioxide Samples

Sample No.	Oxygen-to- uranium ratio	Treatment before analysis
1	2.068	as received
2	2.066	as received
3	2.067	as received
4	2.066	as received
5	2.000	heated for 5 hours in vacuum (10 ⁻⁶ torr) at 1960°C
6	2.012	heated for 5 hours in vacuum (10 ⁻⁶ torr) at 15 4 2°C
7	2.042	heated for 5 hours in vacuum (10 ⁻⁶ torr) at 10 4 6°C

•

TABLE II

Lattice Parameter Measurements of UO2

Run No.	Temperature <u>°C</u>	Lattice Parameter					
1	25	5.469 ± 0.003					
2	1074	5,525					
3	1587	5,566					
4	1415	5.554					
5	886	5.507					
6	925	5.511					
6a	25	5.468					
7	1159	5.533					
8	1244	5.536					
8a	25	5.469					
9	1472	5,558					
10	1301	5.542					
11	1187	5.534					
12	1387	5.548					
13	1542	5.565					
13a	25	5.468					
14	1766	5.584					
15	1822	5.587					
16	1921	5.595					
17	1759	5.580					
18	1670	5.575					
19	1359	5.547					
20	1018	5.522					
21	1825	5.589					
22	870	5.503					
23	1960	5.601					
23a	25	5.470					
24*	845	5.507					
25*	927	5.513					

 Run Nos. 24 and 25 are samples heated to 1960°C in vacuum (10-6 torr) for 5 hours before taking x-ray diffraction patterns.

TABLE III

Lattice Parameter Measurements of ThO_2

Run No.	Temperature °C	Lattice Parameter					
1	25	5.595 ± 0.003					
2	895	5.641					
3	1175	5.655					
4	1280	5.661					
5	1370	5.663					
6	1455	5.671					
7	1665	5.679					
8	1805	5.692					
9	1965	5.703					
10	1025	5.645					
11	2090	5.712					
12	845	5.633					
13	1750	5.687					
14	2025	5.708					
15	1895	5.695					







REFERENCES

- P. Murray and R. W. Thackray, UKAEA Report, AERE-M/M22 (1950).
- I. P. Bell and S. M. Makin, UKAEA Report, RDB(C)TN-70 (1954).
- 3. M. D. Burdick and H. S. Parker, J. Am. Ceram. Soc. <u>39</u>, 181 (1956).
- 4. F. A. Halden, H. C. Wohlers, and R. H. Reinhart, USAEC Report, SRIA-6 (1959).
- 5. J. M. Leblanc and H. Andriesson, EURATOM Report, EURAEC-434 (1962).
- 6. J. A. Christensen, J. Am. Ceram. Soc. 46, 607 (1963).
- 7. J. B. Conway, R. M. Fincel, and R. A. Hein, Third Geneva Conference, CONF-39-50 (1963).
- 8. J. Thewlis, Acta. Cryst. <u>5</u>, 790 (1952).
- 9. F. Gronvold, J. Inorg. Nucl. Chem. 1, 357 (1955).
- 10. D. A. Vaughn, J. R. Bridge, A. G. Wilson, and G. M. Schwartz, Ind. Eng. Chem. <u>49</u>, 1699 (1957).
- 11. C. P. Kempter and R. O. Elliott, J. Chem. Phys. <u>30</u>, 1524 (1959).
- 12. T. W. Baker, UKAEA Report, AERE-M 1624 (1965).
- 13. P. J. Baldock, W. E. Spindler, and T. W. Baker, J. Nucl. Materials <u>18</u>, 305 (1966).
- 14. S. Aronson, E. Cisney, and K. A. Gingerich, J. Am. Ceram. Soc. <u>50</u>, 248 (1967).
- R. F. Geller and P. J. Yavorsky, J. Res. Nat. Bureau of Standards 35, 87 (1945).
- 16. B. Ohnysty and F. K. Rose, J. Am. Ceram. Soc. <u>47</u>, 398 (1964).
- 17. M. Hoch, R. L. Dean, C. K. Hwu, and S. M. Wolosin, Trans. Met. Soc. AIME <u>221</u>, 1162 (1961).
- 18. W. C. Wyder and M. Hoch, Trans. Met. Soc. AIME <u>227</u>, 588 (1963).

- 19. J. B. Nelson and D. P. Riley, Proc. Phys. Soc. (Lond.) <u>57</u>, 160 (1945).
- 20. M. Hoch and F. J. Furman, <u>Thermodynamics</u>, Vol. II, IAEA, Vienna, SM-6613, 517 (1966).
- 21. R. A. Hein, L. H. Sjodahl, and R. Szwarc, J. Nucl. Mater. 25, 99 (1968).
- 22. M. Hoch and H. L. Johnston, J. Phys. Chem. 65, 1184 (1961).
- 23. Thermodynamic and Transport Properties of Uranium Dioxide and Related Phases, IAEA, Vienna, Tech. Rep. 39 (1965).

Security Classification	
DOCUMENT CO	INTROL DATA - RED
ORIGINATING ACTIVITY (Corputete author)	28 REPORT SECURITY CLASSIFICATION
University of Cincinnati	Unclassified
Cincinnati, Ohio 45221	N.A.
REPORT TITLE	
HIGH TEMPERATURE THERMAL EXPAN	SION OF UO2 AND THO2
DESCRIPTIVE NOTES (Type of report and inclusive dates)	
Technical Report - July 1967 th	rough July 1968
Hoch, M., and Momin, A. C.	
REPORT DATE	78 TOTAL NO OF PAGES 76 NO OF REFS
October 1968	17 23
CONTRACT OR GRANT NO F33615-67-0-1565	SE ORIGINATOR'S REPORT NUMBER(5)
6 PROJECT NO	AFML-TR-68-219
7360	
'Task No. 736005	\$5 OTHER REPORT NO(\$) (Any other numbers that may be sealghed this report)
d	None
AVAIL ABILITY/LIMITATION NOTICES	
This document has been approve its distribution is unlimited.	d for public release and sale;
I SUPPLEMENTARY NOTES	12 SPONSORING MILITARY ACTIVITY
None	Air Force Systems Command
	Wright-Patterson AFB, Ohio 45433
3 ABSTRACT	
The thermal expansion of dioxide has been measured betw temperature x-ray diffraction expansion of UO_2 and ThO_2 as m is identical to that obtained Because of this, and because t a rapid increase above $1700^{\circ}C$ be concluded that the major st type disorder. This probably from the tetrahedral into an e	uranium dioxide and thorium een 20 and 2100°C using high techniques. The thermal easured by x-ray diffraction by bulk expansion measurements. he specific heat of UO ₂ shows indicating a disorder, it must ructural defect is a Frenkel involves the oxygen atom moving mpty octahedral position.

.

•

•

Unclassified

Security Classification

14		LINI	K A	LINI	K (B)	LINKC		
	MEY WONDS	AOLE	# T	HOLE	w T	HOLE	W T	
	UO2 Thermal expansion high temperature x-ray diffraction disorder							
	INSTRUCTIONS							

1. ORIGINATING ACTIVITY: Enter the name and address of the contractor, subcontractor, grantee, Department of Defense activity or other organization (corporate author) issuing the report.

2a. REPORT SECURITY CLASSIFICATION: Enter the overall security classification of the report. Indicate whether "Restricted Data" is included. Marking is to be in accordance with appropriate security regulations.

2b GROUP Automatic downgrading is specified in DoD Directive 5200.10 and Armed Forces Industrial Manual. Enter the group number Also, when applicable, show that optional markings have been used for Group 3 and Group 4 as authorized

3. REPORT TITLE: Enter the complete report title in all capital letters. Titles in all cases should be unclassified. If a meaningful title cannot be selected without classification, show title classification in all capitals in parenthesis immediately following the title.

4. DESCRIPTIVE NOTES If appropriate, enter the type of report, e.g., interim, progress, summary, annual, or final. Give the inclusive dates when a specific reporting period is covered.

5. AUTHOR(S) Enter the name(s) of author(s) as shown on or in the report. Enter last name, first name, middle initial. If militars, show rank and branch of service. The name of the principal author is an absolute minimum requirement.

b. REPORT DAT^{*}₂ Enter the date of the report as day, month, year, or month, year. If more than one date appears the report, use date of publication.

TOTAL NUMBER OF PAGES The total page count should follow normal pagination procedures, i.e., enter the number of pages containing information.

TL NUMBER OF REFERENCES. Enter the total number of references cited in the report.

Ball CONTRACT OR GRANT NUMBER of appropriate, enter the applicable number of the contract or grant under which the report was written.

86, α , α , 8d - PROJECT NUMEER. Inter the appropriate military department identification, such as project number, subproject number, system, numbers, task number, etc.

9.a. ORIGINATOR'S REPORT NUMBER(S) Enter the official report number by which the document will be identified and iontrolled by the originating activity. This number must be singue to this report.

In OTHER REPORT NUMBER(5). If the report has been assigned any other report numbers, either by the originator or by the spin originals end or this number(s).

imposed by security classification, using standard statements such as:

- (1) "Qualified requesters may obtain copies of this report from DDC."
- (2) "Foreign announcement and dissemination of this report by DDC is not authorized."
- (3) "U.S. Government agencies may obtain copies of this report directly from DDC. Other qualified DDC users shall request through
- (4) "U. S. military agencies may obtain copies of this report directly from DDC. Other qualified users shall request through.
- (5) "All distribution of this report is controlled. Qualified DDC users shall request through

If the report has been furnished to the Office of Technical Services, Department of Commerce, for sale to the public, indicate this fact and enter the price, if known-

11. SUPPLEMENTARY NOTES: Use for additional explanatory notes.

12. SPONSU ING MILITARY ACTIVITY: Enter the name of the departmental project office or laboratory sponsoring $(paying \ for)$ the research and development. Include address.

13 ABSTRACT. Enter an abstract giving a brief and factual summary of the document indicative of the report, even though it may also appear elsewhere in the body of the technical report. If additional space is required, a continuation sheet shall be attached

It is highly desirable that the abstract of classified reports be unclassified. Each paragraph of the abstract shall end with an indication of the military security classification of the information in the paragraph, represented as (TS) - (S) - (C) - or (U)

There is no limitation on the length of the ubstract. However, the suggested length is from 150 to 225 words.

14 KEY WORDS! Key words are technically meaningful terms or short phrases that characterize a report and may be used as index entries for cataloging the report. Key words must be selected so that no security classification is required. Identifiers, such as equipment model designation, trade name, military project code name, reographic location, may be used as key words but will be followed by an indication of technical context. The assignment of links rules, and weights is optional

Unclassified

Security Classification