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CENTER FOR HIGH ENERGY FORMING

ELEVENTH QUARTERLY REPORT  
OF TECHNICAL PROGRESS

J. D. Mote

April 1, 1968

Army Materials and Mechanics Research Center  
Watertown, Massachusetts 02172

Martin Marietta Corporation  
Denver Division  
Contract DA 19-066-AMC-266(X)  
The University of Denver  
Denver, Colorado

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## ABSTRACT

Strain gage data showing strain- and pressure-time histories versus charge stand-off during explosive deformation is given. The results show a decrease in strain rate with an increase in stand-off at constant charge weight.

Recent results of a numerical solution for deformation and instability in the hydrostatic bulge test are presented.

Summary information is presented for technology transfer; mechanics of energy transfer and high velocity metal deformation; electromagnetic forming; strain rate effects; the effect of explosive forming on the terminal properties of materials, and explosive welding.

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## I. MARTIN MARIETTA CORPORATION

### 1. Strain Gage Measurements

Principal Investigator: G. A. Thurston

#### A. Introduction

A set of tests has been conducted to measure strain- and pressure- time histories during explosive forming. The results show the effect of varying the charge stand-off distance from the blank while holding other forming parameters constant. Table 1 lists parameters in the tests and the four stand-off distances used,  $L = 1, 2, 4$  and  $6$  inches. Nondimensionalized with respect to the die diameter  $D$ , the stand-off ratios were  $L/D = .083, .167, .25,$  and  $.5$ .

Each aluminum blank had three strain gages mounted on the side toward the charge. Strain- time histories were measured for each gage as the blank was free-formed into a vacuum.

The strain- time traces from each gage have been plotted in Fig. 1. The gage locations and orientation are sketched in Fig. 2. Photographs of the oscilloscope traces from each gage are reproduced in Fig. 3. Some gages did not survive the forming event. Those gages that did survive registered maximum strains comparable to the residual strains measured on the surface of the formed dome as shown in Table 1 and Fig. 4. Figures 5 and 6 are plots of strain-rates and the log of strain-rate as a function of the stand-off ratio,  $L/D$ . The plotted strain-rates are the maximum slopes of the strain- time curves.

The results show a decrease in strain-rate with an increase in stand-off distance for a constant charge weight. Past forming experience has shown that an  $L/D$  ratio of  $.167$  leads to uniform draw on free-formed parts. This tendency to pull-in apparently occurs at an early time in the forming because on the curve for  $L = 2, L/D = .167$ , the tangential gage C (see Fig. 1) first records compressive strain, a direct measure of pull-in, and then goes into tension. The same trend starts for  $L = 1$ , but the gage and the blank both failed in this test.

The tendency to pull-in for  $L/D = .167$  was further demonstrated by the final amount of edge draw. This pull-in was limited in these tests to a nominal value of 8 per cent compressive strain by a catch ring that engaged a weld bead on the blank rim. For  $L/D = .167$ , the rim was completely in contact with the ring while uneven draw was exhibited for  $L/D = .5$  and the rim did not completely draw in to the ring. This edge pull-in is tabulated in Fig. 7.

The maximum measured strain-rates due to stretching are on the order of 100 inches per inch per sec which are not extremely high. It should be kept in mind that these strain

TABLE I - Log of Tests and Results

Test No.	Strain Gages	Strain Gage Adhesive (inches)	Standoff Distance (inches)	Strain Gage Results	
055	Bean Co.*	C-2A (Bean Co.)	6	Gage A broke at 3.7% strain and after 1.2 m secs time. Gages B show 2.1% strain unbroken.	
056	Budd Co.*	Narmco	4	Sweep rate for Gage A was too fast to indicate strain. Gage B broke at 4.9% strain and 2.7 m secs. Gage C broke at 2.1% strain and 2.7 m secs.	
057	Budd Co.	GA-4 (Budd Co.)	2	All gages broke early at 60 sec.	
058	Bean Co.	C-2A	4	Gage A broke at 2.8% strain and 1.1 m secs. Gages B show max. strain 3.3% and not broken. Gage C shows max. strain 0.7% and not broken.	
059	Budd Co.	Gage A - GA-4 Gage B & C - Narmco	2	Gage A broke at 4% strain after .35 m secs. time. Gage B broke at .8% strain after .30 m secs. Gage C broke at .8% strain after .75 m secs.	
060	Budd Co.	GA-4	2	Gage A broke at 3.5% strain at .24 m secs. Gage B broke at 3.7% strain at .64 m secs. Gage C broke at 2% strain at .90 m secs.	
066	Bean Co.	Gage A - C-2E Gage B - GA-4 Gage C - GA-4	4	Gage A broke at 3.4% strain at 1.40 m secs. Gage B shows 4.8% strain and was not broken. Gage C shows 1.5% strain and was not broken.	
067	Bean Co.	Gage A - C2-E Gage B - GA-4 Gage C - GA-4	6	Gage A shows 3.4% strain and was not broken. Gage B shows 1.7% strain and was not broken. Gage C shows .35% strain and was not broken.	
068	Bean Co.	Gage A - GA-4 Gage B - GA-4 Gage C - C2-A	1	All gages were broken, also blank was torn. Gage A broke at 4.6% strain and 130 secs. time. Gage B broke at 3.1% strain and 200 secs. time. Gage C broke at .68% strain and 200 secs. time.	

Waterproofing: CW-5 (Budd Co.)  
 Blank Mat'l: AL 2014-G; .050 x 16 3/4" dia.  
 Explosive Charge: Comp A-3; 60 grain  
 Clamp Pressure: 200 psig  
 Pressure Transducer: CR67-265 (Crystal Research Inc.)  
 Distance Charge to Transducer: 4 inches  
 Blasting Cap: 6.9 grain RDX (5 6 gr A-3 equiv.)

\*Bean Co. gages are number EP-08-250 BB-120  
 Budd Co. gages are number HE 141B





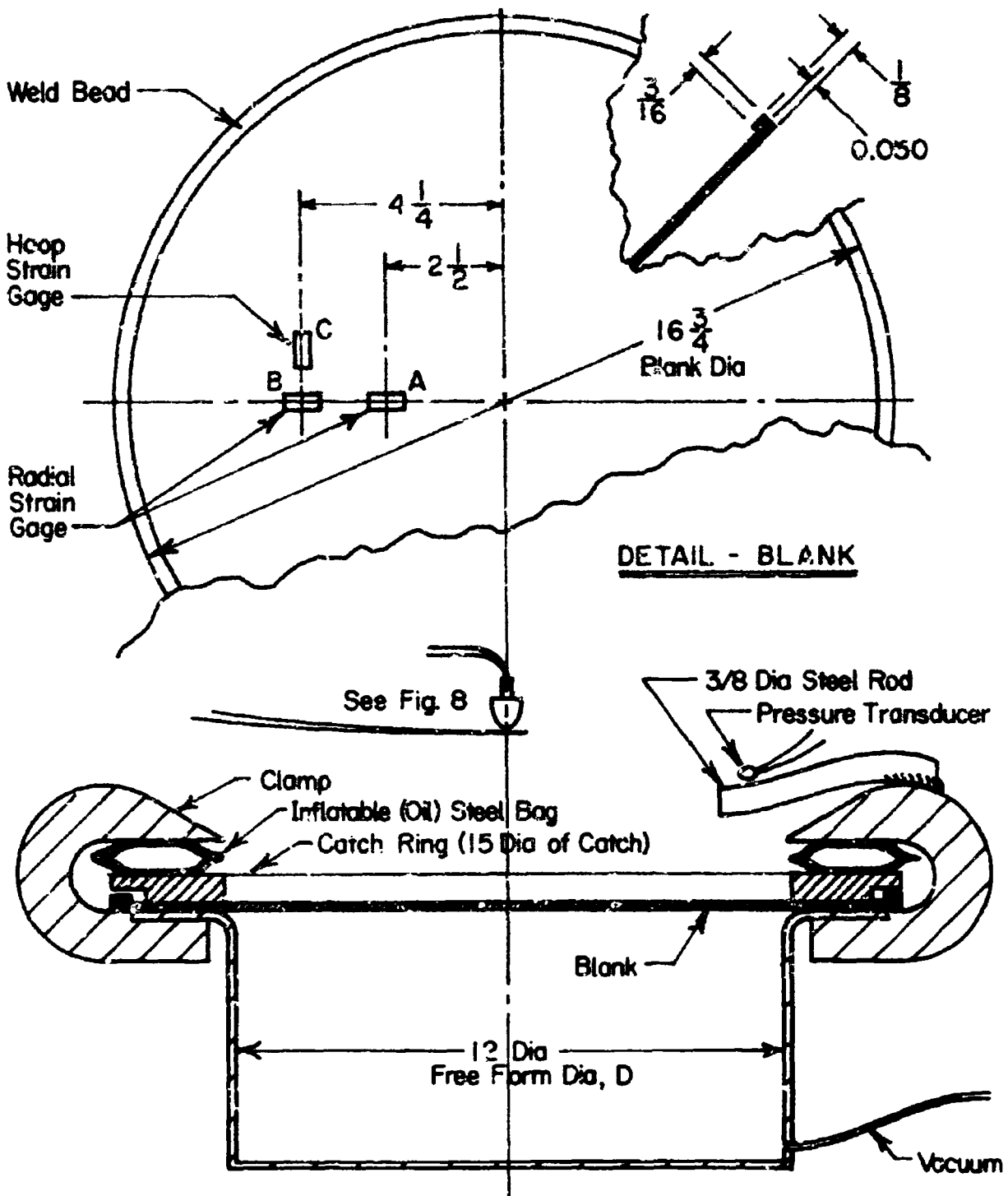


Fig. 2 Schematic of Die and Clamp Set-up

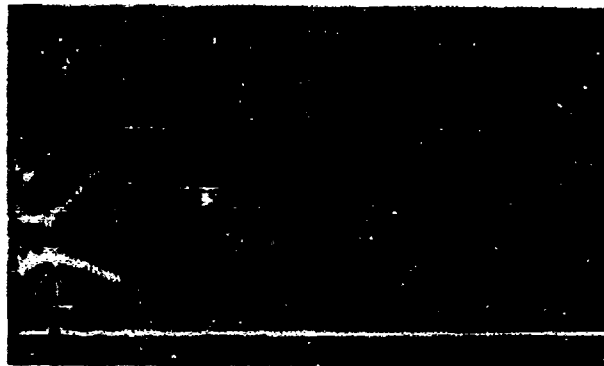
1.73% Strain/cm

Gage A

Gage B

Gage C

Pressure (5,100 psi/cm)



Sweep Rate = 100  $\mu$ secs/cm

Fig. 3a Oscilloscope Trace of Test No. 068.  
Stand-off Distance was 1 Inch

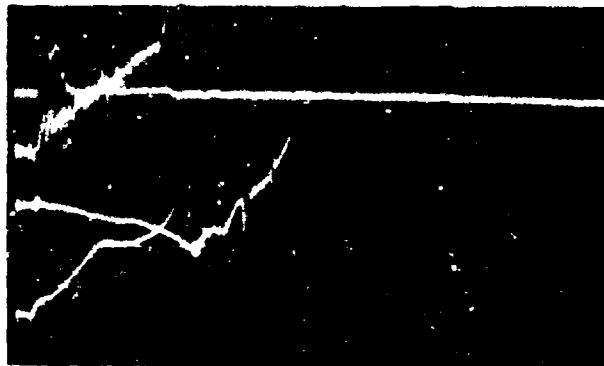
1.73% Strain/cm

Pressure (2,220 psi/cm)

Gage A

Gage C

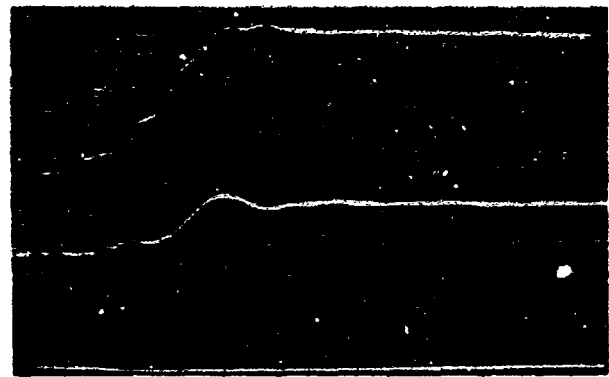
Gage B



Sweep Rate for Pres. & Gage A = 100  $\mu$ secs/cm  
Sweep Rate for Gages B & C = 200  $\mu$ secs/cm

Fig. 3b Oscilloscope Trace of Test No. 060.  
Stand-off Distance was 2 Inches

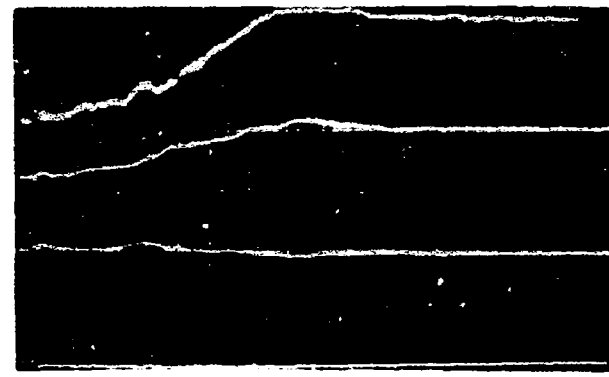
1.73% Strain/cm  
Gage A  
Gage B  
Gage C  
Pressure (5,550 psi/cm)



Sweep Rate = 500  $\mu$ secs/cm

Fig. 3c Oscilloscope Trace of Test No. 066.  
Stand-off Distance was 4 Inches

1.73% Strain/cm  
Gage A  
Gage B  
Gage C  
Pressure (5,550 psi/cm)



Sweep Rate = 500  $\mu$ secs/cm

Fig. 3d Oscilloscope Trace of Test No. 067.  
Stand-off Distance was 6 Inches

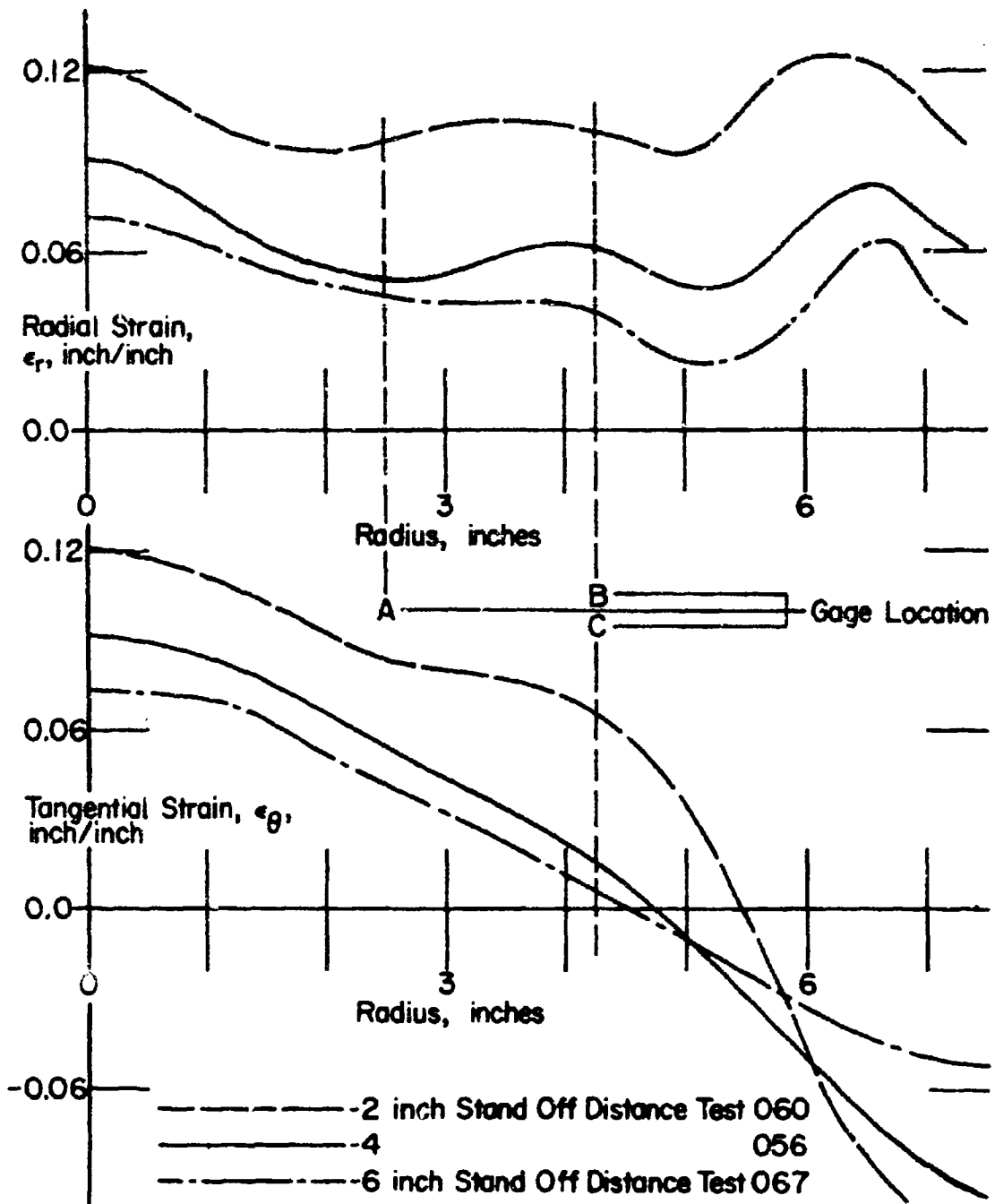


Fig. 4 The Final Strains as a Function of Undeformed Radial Distance

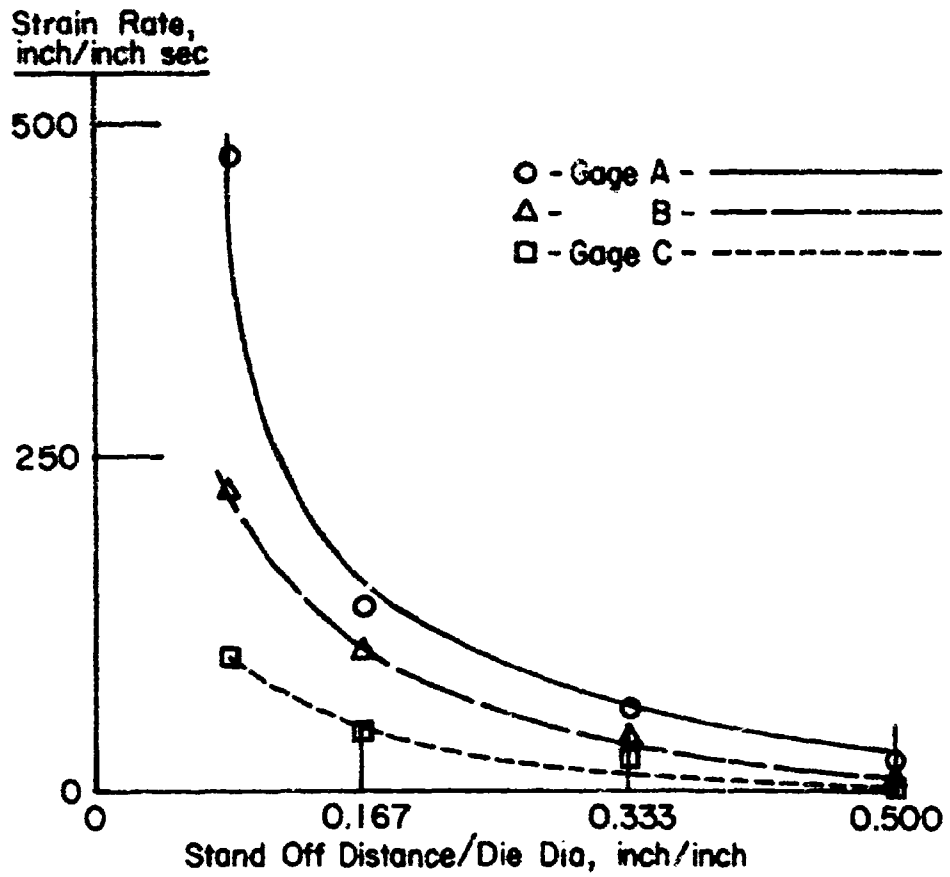


Fig. 5 Strain Rate Plotted as a Function of Stand-off Distance

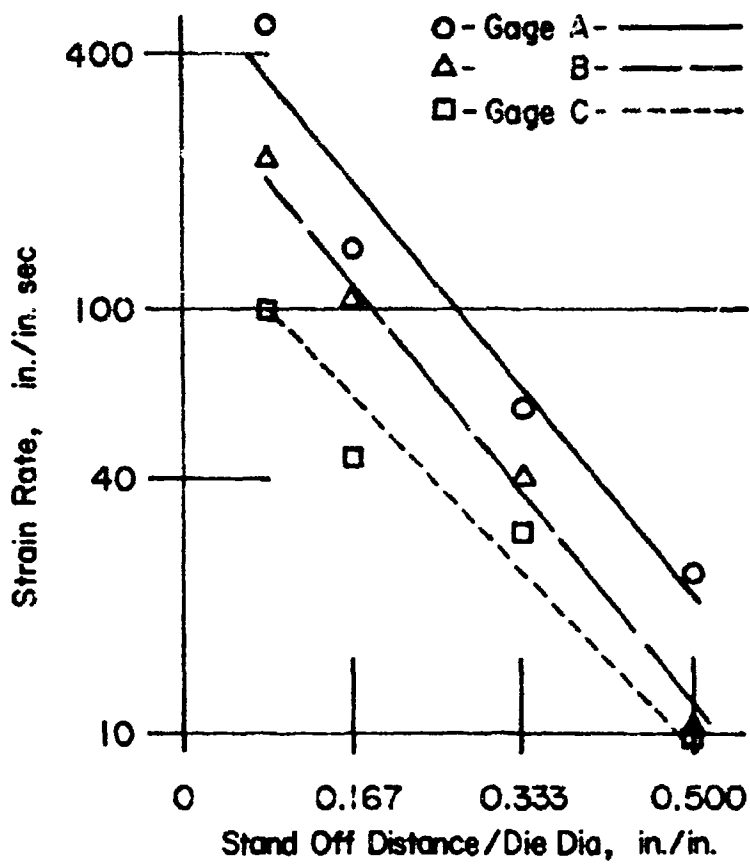
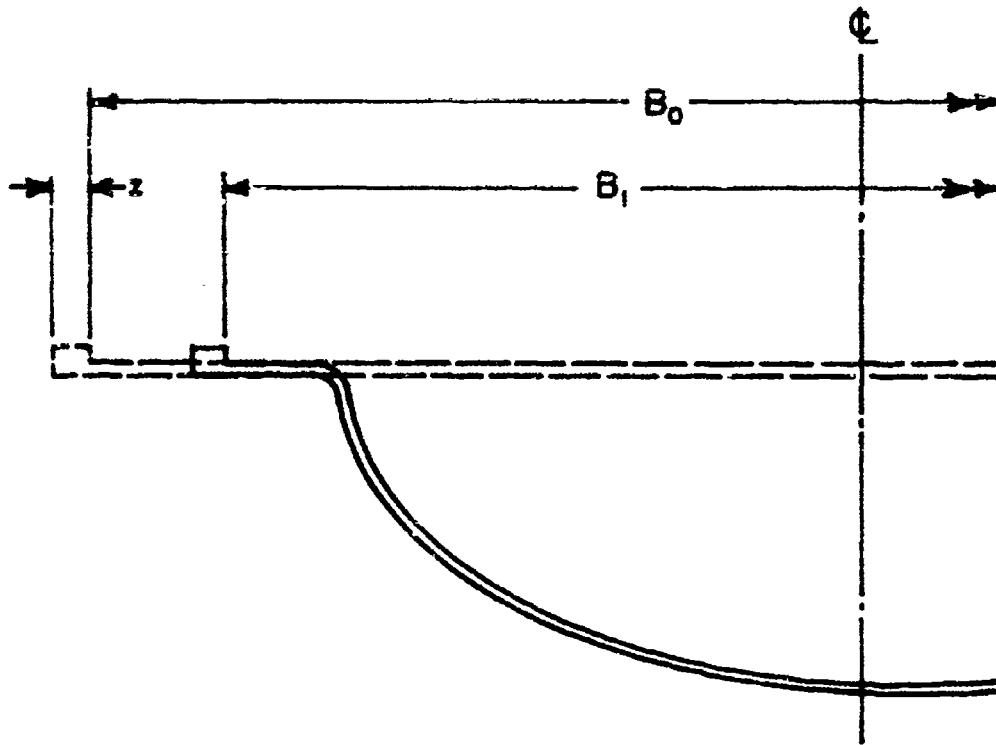


Fig. 6 Strain Rate Plotted as a Function of the Log of Stand-off Distance



Part Test No	$B_0$	$B_1$	$z$	$\epsilon_g @ B_0$
056	16.30	15.00	0.23	$1.30/16.30 = -0.080$
058	16.18	15.10	0.30	$1.06/16.18 = -0.067$
060	16.36	15.02	0.19	$1.34/16.36 = -0.082$
066	16.30	15.40	0.25	$0.90/16.30 = -0.055$
067	16.25	15.50	0.24	$0.75/16.25 = -0.046$

Fig. 7 Tabulated Data Showing Degree of Outside Diameter Pull-in



rates will be lower on large parts for a given L/D ratio because the scaling laws predict equal blank velocities and therefore lower strain rates.

The pressure-time history was measured during each shot by a transducer at the same fixed distance from the charge. The pressure trace at a constant weight of charge proved to be repeatable from test to test. The shock wave traveled at a velocity of Mach 1. The peak pressures from the charge of Composition A-3 proved comparable to the empirical values predicted in the literature for Pentolite.

#### B. General Test Set-up

The blanks to be formed were 2014-O aluminum, .050 inches thick, with a bead welded along its outside diameter. The outside diameter of the blank was nominally 16 3/4 inches. The nominal dimensions of the weld bead are shown in Fig. 2.

A general schematic of the blank clamped to the die can also be seen in Fig. 2. The clamping force upon the blank was applied with hydraulic pressure contained within an inflatable steel bag. This provided uniform clamping pressure against wrinkling all around the blank and was set at 200 psig for all of the tests. A catch ring was used to engage the weld bead during forming to control outside diameter pull-in. A vacuum was drawn within the die cavity and the test was conducted underwater.

The explosive charge was Composition A-3, hemi-spherically shaped and weighting 60 grains. The charge was placed in the center of the blank at the various stand-off distances of 1, 2, 4, and 6 inches and detonated with a Dupont No. 8 blasting cap, which is equivalent to 6.9 grains of PETN.

A 3/8 inch steel rod was extended from the clamp to the middle area of the blank with a pressure transducer mounted to it. The transducer was located between the charge and the rod. Water flow around the rod provides some protection to the piezoelectric pressure transducer by diverting shrapnel from the blasting cap. This resulted in longer life for the pressure transducer without affecting the pressure amplitude.

#### C. Instrumentation System

The test data was recorded on Polaroid Film using a Tektronix Type 555 Dual Beam Oscilloscope with a Dumont Oscilloscope camera. Four channels were available by using CA type amplifiers in a chopped mode. Various sweep rates were tried to determine the coverage of the pressure and strain time event with good resolution. A previously established procedure was followed and consisted of the following:

1. With the scope beams off, record the screen grid on film;

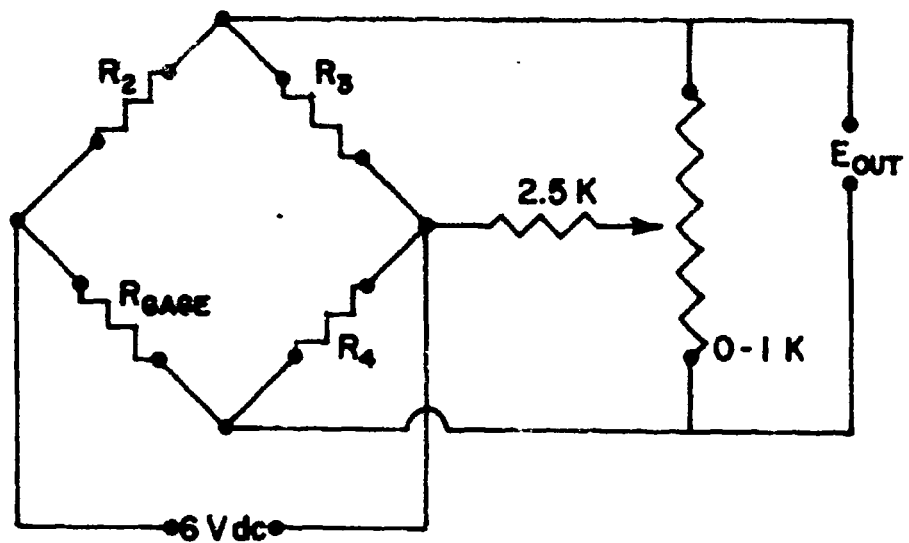
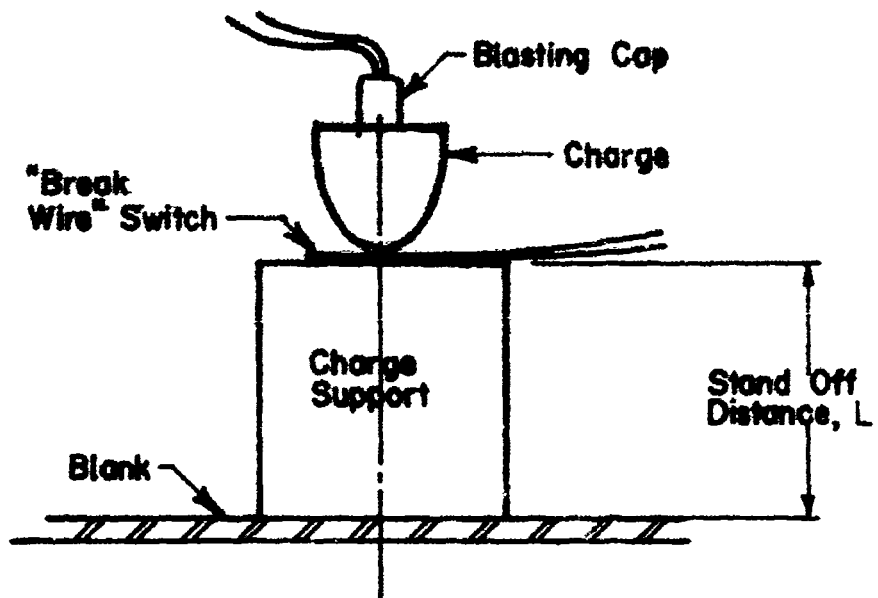


Fig. 9 Strain Gage Bridge Circuit

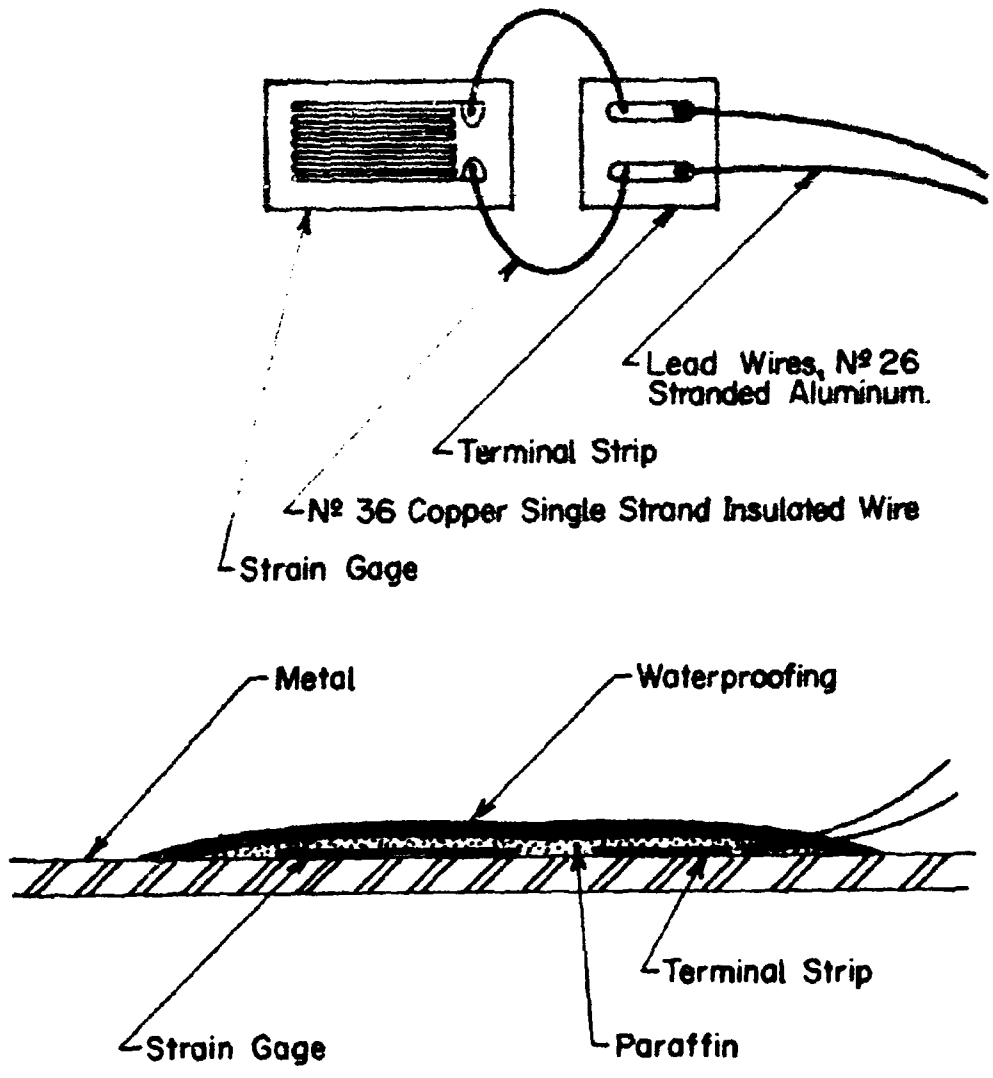


Fig. 10 Strain Gage Installation

2. Set the camera on time, the scope sweep controls on single sweep and external trigger, then open the camera shutter; and,
3. Detonate the charge and close the camera shutter.

The oscilloscope trigger signal was the output of a Wheatstone bridge, excited by a 6 volt battery and originally balanced to 0 volts output. A continuity wire from one bridge leg passed directly beneath the explosive charge as shown in Fig. 8. The blast from the charge broke the wire, produced an instantaneous 2.8 volt bridge output, and triggered the beams of the oscilloscope. Previously, a "strain gage switch" was used to provide the trigger signal. These tests showed both the "strain gage switch" and the "break wire switch" gave an adequate, repeatable, trigger signal to the oscilloscope. The wire is obviously more inexpensive.

Some early tests resulted in the scope beam sweeping much before the time of the pressure or strain event. The trouble was found to be caused by electro-magnetic radiation produced by the Dupont Blasting Machine which was used to detonate the blasting cap. A 6 volt battery was substituted and the problem of an early trigger signal was eliminated.

At times the electrical noise in the system was at high levels. Much of the noise was eliminated by shutting off all induction machinery prior to testing and by tying electrical ground of the water pool with the instrumentation ground. The strain was measured as the output of the strain gage bridge circuit shown in Fig. 9. The excitation voltage was supplied by a 6 volt battery. The three resistor legs of the bridge are strain gages all measuring 120 ohms. All bridges were calibrated for strain by substituting known resistors for the gage.

#### D. Strain Gage Evaluation

Two strain gages measured radial strain; one at  $2\frac{1}{2}$  inches from the blank center and the other at  $4\frac{1}{4}$  inches. The other strain gage measured hoop strain at  $4\frac{1}{4}$  inches from the blank center. A continued evaluation of strain gages and adhesives has resulted in a preference for gages manufactured by William Bean Company. These gages are made of annealed constantan upon a polyimide backing. The polyimide backing is a strong material. Post test examination has not found the backing or the gage to craze as have some other gages. Failure of these strain gages has been isolated to the adhesive bond and to the single strand lead wire connecting the gage to the terminal strip shown in Fig. 10. Microscopic examination of the bonded areas showed separation between the aluminum blank and the cement. While careful metal surface preparation was done prior to applying the gages, the analysis shows that a more elaborate procedure may be needed in this area.

## E. Results of Measurement

The actual film data is shown in Fig. 3 for four tests; one for each of the selected stand-off distances. The strain scales are the same for each test. The sweep rates are not the same in order to provide better resolution for the event and to allow for marginal unknown of the total forming time. The forming plus the relaxation time on some tests was as long as 3 milli-seconds. The broken line of the trace was caused by the chopping of the beam on its horizontal sweep. Some strain histories showed a strain reversal by first measuring compressive strain and latter in time a tensile strain. Strain reversal was particularly true for gage C measuring hoop strain, but could also be seen at gage A for the closer stand-off distances.

Final part contours are shown in Fig. 11. The closer stand-off produces more work on the blank for a constant charge weight.

## 2. Numerical Solution for Deformation and Instability in the Hydrostatic Bulge Test

Principal Investigator: R. Harris

This is a progress report on a master's thesis at the University of Denver. This work by R. Harris has been partially supported by the contract.

Initial results have been obtained from the computer program for predicting strains and displacements in a hydrostatically bulged circular plate specimen. The Tenth Quarterly Report of the Center contains a description of the theory and governing equations. Fig. 12 and 13 show the computed displacements as a function of pressure on annealed copper.

The figures also compare the experimental data and calculations reported by Weil and Newmark\*. At present work is continuing to determine the instability strain at the apex. Weil and Newmark reported experimental instability strains of .430 and .323 while their last converged solution is at  $\epsilon = .215$ . Their extrapolated instability strain was .308. The present computer solution has converged at a strain of .41. A charge of variable must be programmed before the final instability strain at maximum pressure can be calculated.

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\* N. Weil and N. Newmark, "Large Plastic Deformation of Circular Membranes," Journal of Applied Mechanics, Vol. 22, p 533, December 1955

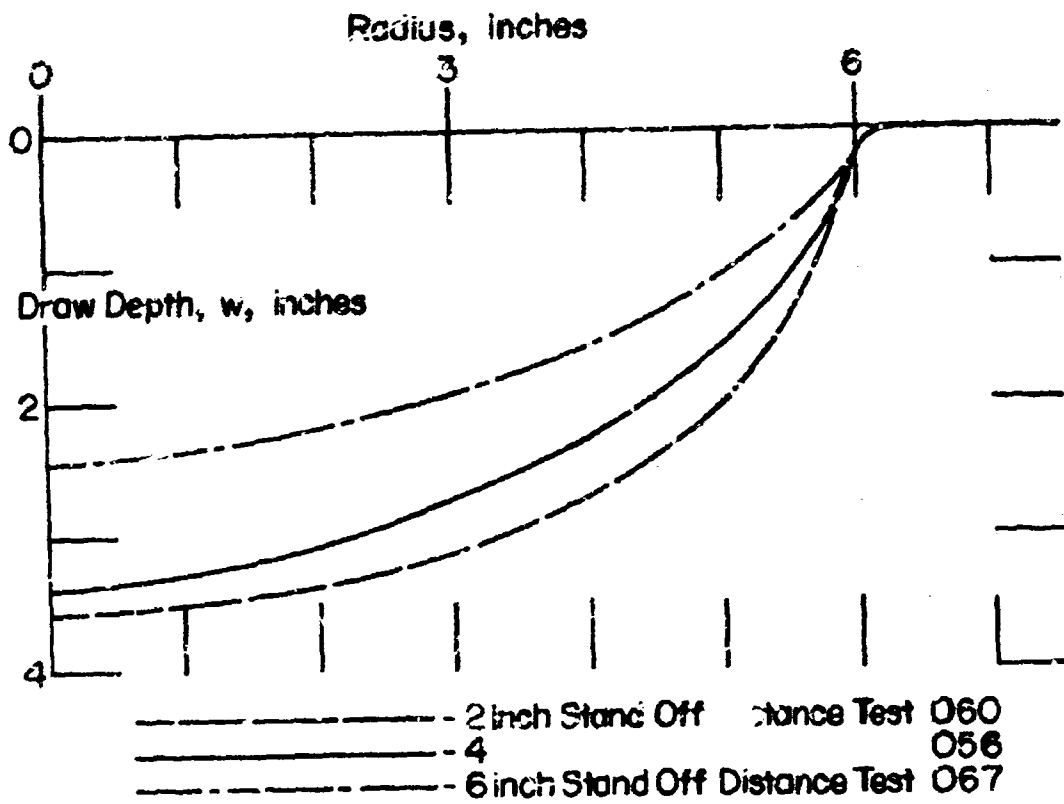


Fig. 11 Plot of Draw Depth as a Function of Blank Radius

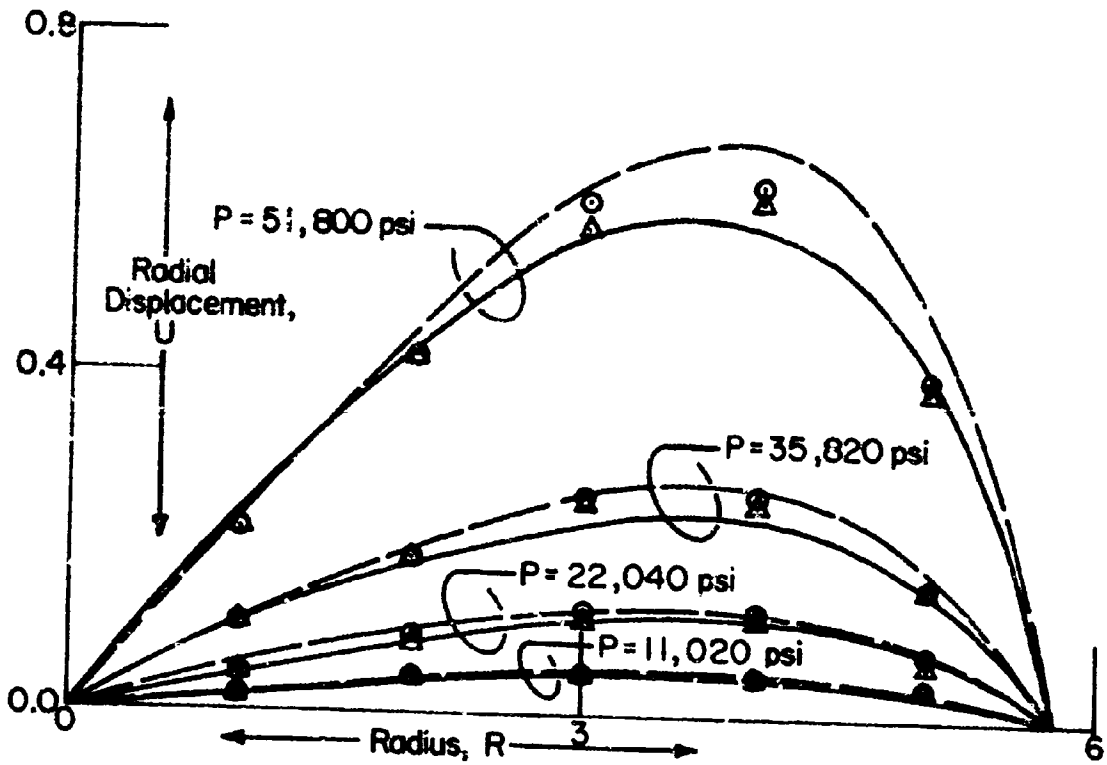


Fig. 12 Radial Displacements as a Function of Radius

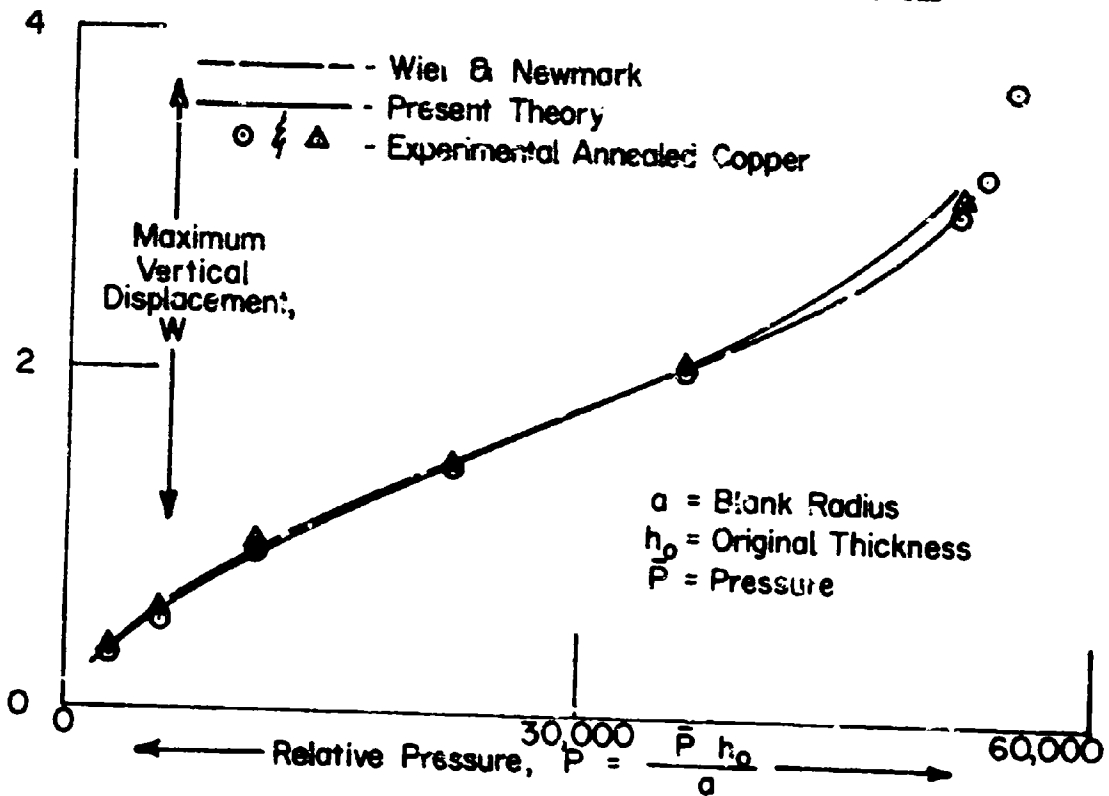


Fig. 13 Polar Displacement as a Function of Radius

W. Johnson\*\* and R. Hill\*\*\* give

$$\epsilon = \frac{m + .5}{2} = \frac{.365 + .5}{2} \frac{8}{11} = .432,$$

$$\epsilon = \frac{8}{11} \left( \frac{m + .5}{2} \right) = \left( \frac{.365 + .5}{2} \right) \frac{8}{11} = .314$$

respectively as their instability strains with  $m$  being the strain hardening exponent.

\*\* W. Johnson, J. L. Duncan, C. Kormi, R. Sowerby, and F. W. Travis, "Some Contributions to High Rate Sheet Metal Forming." Proceedings of the 4th International M.T.D.R. Conference, Pergamon Press, Oxford and New York, 1963.

\*\*\* R. Hill, "A Theory of the Plastic Bulging of a Metal Diaphragm by Lateral Pressure." Phil. Mag., Vol. 41, No. 1133, p 1133-1142, 1950.



## II. UNIVERSITY OF DENVER

### 1. Technology Transfer

Principal Investigator: A. A. Ezra

All the corrected versions of the papers presented at the First International Conference of the Center for High Energy Forming are in the hands of the Publications Department. The Proceedings will be mailed to conference attendees during the first week of July, 1968.

The Second International Conference of the Center for High Energy Forming is planned for June 23-27, 1969, and will be held at the Stanley Hotel, Estes Park, Colorado. Papers are invited for the following sessions:

- A. Explosive Forming
- B. Explosive Welding
- C. Electro-Hydraulic Forming
- D. Electro-Magnetic or Magneto-Motive Forming
- E. Explosive Gas Forming
- F. The Effect of High Energy Rate Forming on Material Properties
- G. The Mechanics of High Energy Rate Forming

Extended abstracts are due October 1, 1968. After being notified of paper acceptance, the manuscript is due February 1, 1969. This will permit the published Proceedings to be available at the Conference.

Professor Ezra is giving a course in High Energy Mechanics during the Spring Quarter at the University of Denver. The results of research programs of the Center will be presented as part of the course.

A special report on Shock Hardening of Metals has been completed. It will be published and made available after August 1, 1968. Another special report on the Mechanics of Energy Transfer in Explosive Forming has also been completed and will be made available after August 1, 1968. A report on the shock twinning behavior of molybdenum is in preparation.

### 2. Mechanics of Energy Transfer and High Velocity Metal Deformation

Principal Investigator: John A. Weese  
Graduate Students: Michael Malcolm, Larry Ching

Computer programs for the reduction of data from the experiments to determine underwater explosion constants for pentolite have been written and preliminary results obtained. The programs are now being refined and an investigation is underway to correlate the results with those obtained earlier by other investigators. Methods for predicting the final deformation of explosively formed rings have been modified and improved. Computer programs have been written for utilizing

strain gage readings taken while boring out an explosively bulged thickwalled tube to determine its state of residual stress. Associated with this, an analytical study is being conducted on the response of cylindrical dies to explosive loading.

### 3. Electromagnetic Forming

Principal Investigator: W. N. Lawrence  
Graduate Student: L. Gilbert

During the first quarter of 1968, the mechanical assembly of one half of the capacitor bank facility was completed. Each component of the bank has been tested for high voltage integrity ('hi-pot' tested) to 30,000 volts, which is 50% above the capacitor rating of 20,000 volts. Corona has been held to a minimum through careful design of connections. All ignitrons have been individually tested. In addition, several magnetic forming coils have been constructed and one was used in the assembly of the bank to install the ignitron in the ignitron shielding container. Some effort was put into developing a theoretical basis for the energy transfer and design limitations of the forming coil. In addition, the Center was contacted concerning the feasibility of swaging rotating bands on large artillery shells by magnetic forming and a small scale test evaluation was made for the Naval Ordnance Test Station at Louisville, Kentucky.

### 4. Strain Rate Effects

Principal Investigator: C. Hoggatt

During this report period curve fitting techniques were re-evaluated to facilitate reduction of the displacement-time data of freely expanding rings into dynamic stress-strain rate relationships. Early in this investigation it was found that all of the displacement-time histories of the expanding ring tests obeyed parabolic relationships out to approximately 10% strain which is the maximum strain achieved to date. Maximum deviation of the recorded data from parabolic behavior was normally around 0.0005 inches at any position on the curve. This held true for all three materials tested to date, 6061-T6 aluminum, 1010 mild steel and 6Al-4V-titanium. It should be noted that these materials represent three different crystal structures (face centered cubic, body centered cubic and hexagonal close packed respectively); three ranges of material density ( $.25 \times 10^{-3}$ ,  $.733 \times 10^{-3}$  and  $.414 \times 10^{-3}$  lb-sec<sup>2</sup>/in<sup>4</sup> respectively); and three ranges of dynamic material strength (45, 100 and 230 ksi respectively.) In addition, 6061-T6 aluminum exhibits no appreciable increase in dynamic flow stress over static behavior while the other two materials exhibit significant increases in the flow stress over static values. However, even with such a wide variance of conditions all investigations to date indicate that the parabolic relation for displacement-time behavior holds for all materials tested, and that the dynamic stress-strain rate behavior obtained from initial tests for these materials agrees very well with dynamic relationships published by Maiden (1). A few additional tests are required to fill in existing gaps in the data to finalize the dynamic relationships for these three materials. Therefore, complete stress-strain strain rate relationships will be presented

in the next quarterly report.

It is presently anticipated that the dynamic uniaxial behavior of 2014 and 2024 aluminum alloys will also be determined during the next report period.

5. Comparison of Terminal Properties of Explosively and Conventionally Formed Aluminum Alloy 2014

Principal Investigators: H. Otto and N. Crava  
Graduate Students: R. Mikesell, D. Olsen, S. Shen

For this program, free formed configurations were selected to eliminate the parameter of die contact. A shock situation does occur upon die contact with explosively formed material which could influence any comparison. Twelve inch domes of both annealed (O) and solution heat treated and artificially aged (T6) stock of 2014 were both explosively and rubber press formed. The die configuration was exactly the same in both instances. Prior to forming, the blanks were gridded with photo-resist so the strain could be determined at any point after forming.

Strain measurements after forming were conducted so fatigue and stress corrosion specimens with comparable strains could be cut from the domes formed by both methods. Although the radius of curvature is a problem in specimen selection, there are relatively flat portions in the dome with comparable strains from dome to dome. The fatigue specimens were cut so these portions corresponded to the gage length (1.2 inches) of the tensile fatigue specimens. The gripping surfaces were flattened so as not to disturb the metal through the gage length. The specimens are electropolished after machining to remove any surface imperfections that could act as stress raisers during testing. Properties of explosively and conventionally formed materials will be measured both after forming and after heat treatment.

The fatigue tests are being conducted at a rate of 15 cps. Cyclic tensile-tensile loads are being established so failures will occur in the as-received stock at  $2 \times 10^6$  cycles or less. Since the fatigue properties are to be comparative based on the type of forming, it is not appropriate to establish a conventional fatigue stress at this time.

Stress corrosion specimens measuring  $1/2 \times 2-1/4$  inches were taken from several areas of the domes. The curvature of these specimens does not hinder testing. Specimens with comparable residual strains were again selected from domes formed by both methods. These specimens are being tested in 3.5% NaCl solution under continuous immersion. Tests are being conducted in both the as-formed and at a stress of 75% of the "as-received" yield strength. The stressed specimens are held in a 3 point loading test fixture with a 1.7 inch span. No cracks have been observed in the explosively formed 2014-O after 15 days exposure.

In addition to continuous immersion tests, intermittent immersion will also be used. Facilities are being adapted for the intermittent tests.

Mr. Mikesell has been concerned with the modification of the TM-6 Marquardt Universal Testing Machine for adaptation to fatigue testings. By using a sine wave generator, an alternating load can be applied. Actual calibration requires the use of an oscilloscope so the amplitude can be adjusted correctly.

Mr. Olsen has successfully prepared electron micrographs of 2014-O and 2024-O aluminum alloys. Studies with the 2014-O will be used to develop heat treatment schedules for this material for the comparative studies of explosive and rubber press formed specimens.

Mr. Shen has been concerned with stored energy studies and the determination of strain in the domes. For the former, a tin-solution calorimeter has been used to measure the stored energy in cold-worked copper. Copper was selected for this feasibility study, since the amount of stored energy as a function of cold work is relatively high. Owing to the lack of sensitivity and capacity of the calorimeter, it was not possible to detect the stored energy in the copper, even though it was severely cold worked.

#### 6. Shock-Twinning Behavior of Molybdenum

Principal Investigator: Dr. S. Mahajan

It was indicated in the previous Quarterly Report that as-received molybdenum strips would be annealed to obtain coarse grained samples. This treatment would have increased the frequency of shock-twinning, as observed on the macroscopic scale. However, the attempts to produce coarse grained molybdenum were not successful because of the embrittlement problems. For example, the samples annealed at 1500°C for 1 and 4 hours in vacuum could be broken by pressing them between finger nails.

A technique has been developed for preparing thin foils from molybdenum strips. The samples are first chemically thinned in a solution containing HF and HNO<sub>3</sub> acids. Chemically thinning operation is followed by electropolishing in an electrolyte containing 25cc of H<sub>2</sub>SO<sub>4</sub> and 75cc of C<sub>2</sub>H<sub>5</sub>OH. Sections suitable for transmission electron microscopy are cut from the thin areas.

As-received, as-received + shocked, and 6% prestrain + shocked samples of the fine grain stock were examined by transmission electron microscopy; the shock pressure used was 90 kbars. Following are the tentative conclusions:

- A. Markings resembling twins have been observed in as-received-shocked samples; no such markings have been observed in the pre-strained-shocked samples. No definite conclusion can be drawn about these markings until the diffraction patterns have been analyzed.

- B. Dislocation densities in as-received-shocked and 6% prestrained-shocked samples are almost identical, implying that initial mobile dislocation density does not have much effect on the dislocation density observed after shocking.
- C. In the shocked material, dislocations are distributed uniformly throughout the matrix; there was no indication of cell formation. Dislocations, however, are highly cusped and jogged. A few dislocations loops are also observed.

7. Properties of Explosively Formed AISI Steels 4130, 4140, and 4340

Principal Investigator: H. Otto  
Graduate Student: R. Mikesell

A flat bottom die is currently being constructed for this phase of the work. A combination holddown ring and blank clamping fixture has been completed. The die is designed to accommodate steel with either high or low ductility. Steels have been ordered for this phase and should be available when the die is completed. Fatigue and fracture toughness tests will be conducted on the steel in the "as-formed" and "as-formed and heat treated" conditions.

8. Explosive Welding

Principal Investigator: S. H. Carpenter  
Graduate Students: A. Paddock, D. Olsen

During the last quarter numerous explosive welds of various material combinations and geometries were made at the request of parties and organizations interested in the explosive welding process. A weld for the NASA-Lewis Laboratory was particularly interesting. Iron and tantalum foils were explosively welded to a tungsten-25% rhenium alloy giving satisfactory bonding on the first attempt. The welding of Ti-6Al-4V has led to a spin off contract from the Boeing Company.

Current work is now being directed toward a better metallurgical understanding and characterization of the weld interface. Mr. A. Paddock is studying Cu-Ag welds following various heat treatments using the electron microprobe. With this study Mr. Paddock hopes to determine any anomalies in composition and also study diffusion processes in and along the interface. Mr. Olsen hopes to study the weld interface by means of a transmission electron microscopy. This study has moved slowly due to the difficult polishing procedures necessary to isolate the weld interface. He is currently building a jet polisher which will greatly expedite the polishing process.

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