

Date 7/23/80

To: Micrographics Division, Office of Publications.

From: Special Reproduction Unit, NTIS.

Please process the attached document, ACCESSION NUMBER AD-609 132,  
as circled below:

1. Fiche for file.  
Return document to Special Reproduction Unit, NTIS
2. Fiche for file.  
Fill orders.  
Return document to Special Reproduction Unit, NTIS
3. Fiche for file.  
Return document to Special Reproduction Unit, NTIS,  
ATTN: \_\_\_\_\_ *Susie K.*
4. Fiche for file.  
Return document to Special Reproduction Unit, NTIS,  
ATTN: \_\_\_\_\_  
Printing requests are pending.

---

---

To: Special Reproduction Unit, NTIS.

From: Micrographics Division, Office of Publications.

The attached document has been \_\_\_\_\_  
*AD-609 132*

AD-609 132

ELECTRODEPOSITED, ELECTROLESS, AND ANODIZED  
COATINGS ON BERYLLIUM

Battelle Memorial Institute  
Columbus, Ohio

September 1964

DA AD 609132

September 1, 1964  
DMIC Memorandum 197

ELECTRODEPOSITED, ELECTROLESS, AND ANODIZED  
COATINGS ON BERYLLIUM

DEFENSE METALS INFORMATION CENTER  
BATTELLE MEMORIAL INSTITUTE  
COLUMBUS, OHIO 43201

Unclassified

Security Classification

DOCUMENT CONTROL DATA - R&D

(Security classification of title, body of abstract and indexing annotation must be entered when the overall report is classified)

1. ORIGINATING ACTIVITY (Corporate author) Battelle Memorial Institute Defense Metals Information Center 505 King Avenue, Columbus, Ohio 43201		2a. REPORT SECURITY CLASSIFICATION Unclassified	
		2b. GROUP	
3. REPORT TITLE Electrodeposited, Electroless, and Anodized Coatings on Beryllium			
4. DESCRIPTIVE NOTES (Type of report and inclusive dates) DMIC Memorandum			
5. AUTHOR(S) (Last name, first name, initial) Beach, John G.			
6. REPORT DATE September 1, 1964		7a. TOTAL NO OF PAGES 5	7b. NO. OF REFS 26
8a. CONTRACT OR GRANT NO AF 33(615)-1121		9a. ORIGINATOR'S REPORT NUMBER(S) DMIC Memorandum 197	
b. PROJECT NO 8975		9b. OTHER REPORT NO(S) (Any other numbers that may be assigned this report)	
c.		d.	
10. AVAILABILITY/LIMITATION NOTICES Copies may be obtained from DMIC at no cost by Government agencies and by Government contractors, subcontractors, and their suppliers. Others may obtain copies from the Office of Technical Services, U. S. Department of Commerce, Washington, D. C. 20230			
11. SUPPLEMENTARY NOTES		12. SPONSORING MILITARY ACTIVITY United States Air Force, Research and Technology Division, Wright-Patterson Air Force Base, Ohio 45433	
13. ABSTRACT Because of expanding interest in the uses for beryllium, the Defense Metals Information Center has summarized information on surface finishing and coating of this metal. Its light weight and high modulus position beryllium as a favorable structural material for aerospace needs. The performance of beryllium is benefitted by preferred surface finishing and coatings. Available information is presented on electroless, electrodeposited and anodized coatings on beryllium. (Author{}			

DD FORM 1 JAN 64 1473

Unclassified  
Security Classification

14	KEY WORDS	LINK A		LINK B		LINK C	
		ROLE	WT	ROLE	WT	ROLE	WT
	Beryllium Coatings Electrodeposited coating Electroless coating Anodized coating Cleaning Surface Finishing Chemical polishing						

INSTRUCTIONS

1. **ORIGINATING ACTIVITY:** Enter the name and address of the contractor, subcontractor, grantee, Department of Defense activity or other organization (*corporate author*) issuing the report.
- 2a. **REPORT SECURITY CLASSIFICATION:** Enter the overall security classification of the report. Indicate whether "Restricted Data" is included. Marking is to be in accordance with appropriate security regulations.
- 2b. **GROUP:** Automatic downgrading is specified in DoD Directive 5200.10 and Armed Forces Industrial Manual. Enter the group number. Also, when applicable, show that optional markings have been used for Group 3 and Group 4 as authorized.
3. **REPORT TITLE:** Enter the complete report title in all capital letters. Titles in all cases should be unclassified. If a meaningful title cannot be selected without classification, show title classification in all capitals in parenthesis immediately following the title.
4. **DESCRIPTIVE NOTES:** If appropriate, enter the type of report, e.g., interim, progress, summary, annual, or final. Give the inclusive dates when a specific reporting period is covered.
5. **AUTHOR(S):** Enter the name(s) of author(s) as shown on or in the report. Enter last name, first name, middle initial. If military, show rank and branch of service. The name of the principal author is an absolute minimum requirement.
6. **REPORT DATE:** Enter the date of the report as day, month, year, or month, year. If more than one date appears on the report, use date of publication.
- 7a. **TOTAL NUMBER OF PAGES:** The total page count should follow normal pagination procedures, i.e., enter the number of pages containing information.
- 7b. **NUMBER OF REFERENCES:** Enter the total number of references cited in the report.
- 8a. **CONTRACT OR GRANT NUMBER:** If appropriate, enter the applicable number of the contract or grant under which the report was written.
- 8b, 8c, & 8d. **PROJECT NUMBER:** Enter the appropriate military department identification, such as project number, subproject number, system numbers, task number, etc.
- 9a. **ORIGINATOR'S REPORT NUMBER(S):** Enter the official report number by which the document will be identified and controlled by the originating activity. This number must be unique to this report.
- 9b. **OTHER REPORT NUMBER(S):** If the report has been assigned any other report numbers (*either by the originator or by the sponsor*), also enter this number(s).
10. **AVAILABILITY/LIMITATION NOTICES:** Enter any limitations on further dissemination of the report, other than those

imposed by security classification, using standard statements such as:

- (.) "Qualified requesters may obtain copies of this report from DDC."
- (2) "Foreign announcement and dissemination of this report by DDC is not authorized."
- (3) "U. S. Government agencies may obtain copies of this report directly from DDC. Other qualified DDC users shall request through \_\_\_\_\_."
- (4) "U. S. military agencies may obtain copies of this report directly from DDC. Other qualified users shall request through \_\_\_\_\_."
- (5) "All distribution of this report is controlled. Qualified DDC users shall request through \_\_\_\_\_."

If the report has been furnished to the Office of Technical Services, Department of Commerce, for sale to the public, indicate this fact and enter the price, if known.

11. **SUPPLEMENTARY NOTES:** Use for additional explanatory notes.
12. **SPONSORING MILITARY ACTIVITY:** Enter the name of the departmental project office or laboratory sponsoring (*paying for*) the research and development. Include address.
13. **ABSTRACT:** Enter an abstract giving a brief and factual summary of the document indicative of the report, even though it may also appear elsewhere in the body of the technical report. If additional space is required, a continuation sheet shall be attached.

It is highly desirable that the abstract of classified reports be unclassified. Each paragraph of the abstract shall end with an indication of the military security classification of the information in the paragraph, represented as (TS), (S), (C), or (U).

There is no limitation on the length of the abstract. However, the suggested length is from 150 to 225 words.

14. **KEY WORDS:** Key words are technically meaningful terms or short phrases that characterize a report and may be used as index entries for cataloging the report. Key words must be selected so that no security classification is required. Identifiers, such as equipment model designation, trade name, military project code name, geographic location, may be used as key words but will be followed by an indication of technical content. The assignment of links, rules, and weights is optional.

TABLE OF CONTENTS

	<u>Page</u>
SUMMARY . . . . .	1
INTRODUCTION. . . . .	1
CLEANING AND SURFACE FINISHING. . . . .	1
COATINGS ON BERYLLIUM . . . . .	2
Electrodeposited Coatings. . . . .	2
Precleaning for Activation and Plating. . . . .	2
Zinc-Immersion Method . . . . .	3
Direct Plating Method I . . . . .	3
Direct Plating Method II. . . . .	4
Electroless Coatings . . . . .	4
Electroless Nickel. . . . .	4
Electroless Platinum. . . . .	4
Anodized Coatings. . . . .	4
REFERENCES. . . . .	5

# ELECTRODEPOSITED, ELECTROLESS, AND ANODIZED COATINGS ON BERYLLIUM

John G. Beach\*

## SUMMARY

Coatings are applied to beryllium for four principal reasons:

- (1) To resist corrosion and oxidation
- (2) To facilitate joining
- (3) To improve resistance to wear
- (4) To increase thermal emissivity.

Nonmetallic coatings are produced by oxidation at high temperature or by electrochemical anodic treatment. Metallic coatings are produced by electroplating, electroless plating, flame spraying, and metallurgical cladding.

## INTRODUCTION

This memorandum was prepared to bring together information made available to DMIC on the prefinishing and coating of beryllium by electroplating, electroless plating, and anodizing, used to provide both metallic and nonmetallic coatings.

Because of its light weight, high modulus of elasticity, and good thermal and nuclear properties, beryllium is of interest as a structural material in aerospace systems, as a component of gyroscopes, and as components in nuclear systems. To obtain efficient use of beryllium in these applications, however, coatings often are needed.

Beryllium metal forms a "protective" oxide film with a limiting thickness of about 100 Å ( $10^{-6}$  cm) when exposed to air at room temperature for about 2 hours. Beryllium metal resists attack by dry air, nitrogen, hydrogen, and carbon dioxide at temperatures up to 1500 F. It is also resistant to pure water (free of halogen ions) and liquid metals (free of oxygen).

Beryllium, however, is corroded by water at high temperature, such as may be used in nuclear systems, and by oxygen in liquid metals such as sodium, which is used for heat transfer, unless the oxygen content is maintained at a very low level (<10 ppm). Beryllium is corroded by the fluorinated oils used in gyroscopes. It also is corroded by water containing as little as 0.5 ppm of chloride (considerably less chloride than that found in city or river water), and it pits in sea water. Beryllium has lower emissivity than is desired in space systems. The performance of beryllium in all of these environments can be improved by the application of coatings. For joining by soldering or by pressure bonding at moderately low temperatures, it is necessary to coat beryllium with a suitable metal. Increased wear resistance also can be obtained by coatings on beryllium.

There is, therefore, considerable need to develop coatings of various types for beryllium. Work no doubt will be continued on such problems, and additional information can be expected in the future.

Proper machining techniques<sup>(1)</sup> and safe handling procedures<sup>(2)</sup> should be observed with beryllium.

\* Senior Chemical Engineer, Electrochemical Engineering, Battelle Memorial Institute.

## CLEANING AND SURFACE FINISHING

Unexplained corrosion of beryllium components can often be traced to bad practice in the manufacture of hardware.<sup>(3)</sup> Since the metal is extremely sensitive to corrosion by chlorides, surface cleanliness is extremely important. Detergent cleaning followed by distilled-water rinsing and drying can minimize corrosion of machine parts, which is attributed to a chloride-salt residue from tap water and/or "fingerprinting". Machined parts that were cleaned prior to exposure to one cycle of the MIL-E-5272 humidity test\* showed little or no evidence of corrosion. Parts that were not cleaned showed considerable localized corrosion.<sup>(3)</sup>

Damaged metal in the surface layer of machined beryllium consists of microcracks and twinned areas. Although heat treatment can remove the effect of twinning damage on the mechanical properties of beryllium, it does not remove cracks, which will retain fluids and contaminants.

Mechanical lapping, if carried far enough, can overcome the microcrack problem. Smearing of the surface metal during machining or lapping must be avoided or contaminants will be entrapped. Diamond lapping compounds are recommended for beryllium. Generally, the removal of 2 mils of a machined surface is required to remove microcracks.

Chemical polishing of machined beryllium is effective in revealing and removing microcracks along with the twinned metal structure. However, chemical polishing will also selectively remove oxide particles and certain inclusions from the beryllium surface and thereby produce a pitted surface which at times can be less desirable than the microcracks and twins in the surface layers.

There is little doubt that, if maximum properties of beryllium parts are desired, cleaning and surface finishing should be included in the design specifications. Treatments that can be recommended for maximum properties include finishing cuts of 0.002 inch by machining followed by chemical etching or polishing to remove an additional 0.002 inch of the surface layer. Unless subsurface cracking is present, the above treatments can provide a crack-free beryllium surface.

Annealing at 1450 F and furnace cooling and/or chemical removal of the machine-damaged surface are effective methods to obtain maximum properties in beryllium.<sup>(4)</sup> However, in the case of highly wrought forms, such as extrusions or sheet, difficulties with stress relieving through twin formation are accentuated by annealing because of the formation of additional twins.<sup>(5)</sup>

The effects of surface condition on the hot-pressed properties of beryllium are illustrated by data on the following page.

Other data for beryllium that was originally sawed from a hot-pressed block show greater room-temperature tensile strength by more than 40 per cent, suggesting differences in material, the processing, and/or the testing procedures.<sup>(5)</sup>

\* Two hours' heating to temperature, 6 hours' exposure at 170 F to 100 per cent relative humidity, and 16 hours' cool-down and exposure to condensed moisture.

Room-Temperature Tensile Data<sup>(4)</sup>

Treatment of Beryllium (Y-5802)	UTS, YS, ksi		Contraction, %
	UTS, ksi	YS, ksi	
I. Mechanical polish	31.5	25.4	0.2
II. I plus chemical polish	31.6	25.4	0.4
III. I plus vacuum anneal	31.5	26.4	2.0
IV. II plus vacuum anneal	31.7	25.8	1.7

1200 F Tensile Data<sup>(4)</sup>

Treatment of Beryllium (Y-6825)	UTS, YS, ksi		Contraction, %
	UTS, ksi	YS, ksi	
V. As-machined	16.6	14.9	2.9
VI. Chemical polish	16.9	15.8	7.6
VII. Vacuum anneal	20.1	17.7	15.0
VIII. VI plus VII	19.7	17.0	13.1

Room-Temperature Tensile-Impact Data<sup>(4)</sup>

Treatment of Beryllium (Y-4540)	Ft-lb at 11 ft/sec		Ft-lb at 17 ft/sec
	at 11 ft/sec	at 17 ft/sec	
Same as I above	1.3	0.2	
Same as II above	4.5	3.2	
Same as IV above	4.7	-	

Reference 5

Treatment of Beryllium	UTS, YS, ksi		Elongation, % in 1 inch
	UTS, ksi	YS, ksi	
V. As machined	45	36	1.0
IX. Ground (-2 mils)	51	39	1.3
X. Ground (-5 mils)	50	37	1.3
XI. V plus anneal	51	34	2.7
XII. IX plus anneal	54	36	3.0
XIII. X plus anneal	51	35	2.7
XIV. XI plus chemical etch	50	35	2.2
XV. XII plus chemical etch	53	35	3.2
XVI. XIII plus chemical etch	50	34	2.2

A beneficial effect of carefully removing the machine-damaged metal, 2 to 5 mils of the surface, is obvious from these data as a 50 per cent increase in the ultimate tensile strength. A beneficial effect of annealing to remove surface twins is also apparent, as an increase of about 100 per cent in ductility.

The effects of chemical surface finishing on the properties of beryllium and the performance of beryllium parts are not obvious, often because of the overriding effects that are attributed to the prior history of the part. In addition, the type of finishing solution and the operating conditions, along with the amount of metal removed, have been varied; thus, chemical-finishing effects are difficult to resolve with the available data.

COATINGS ON BERYLLIUM

Electrodeposited Coatings

Adherent electroplating on beryllium involves specific pretreatments of the metal surface.<sup>(4,6-15)</sup> Beryllium surfaces that are to be coated usually have been machined; therefore, gross scale and sur-

face impurities are not a primary problem. If needed, the surface can be descaled by pickling in a hydrofluoric-nitric acid solution (2 vol % of 46% HF, 50 vol % of 70% HNO<sub>3</sub>, and 48 vol % of water at room temperature).

Pretreatment of beryllium for adherent plating includes several precleaning and activation steps along with intermediate water rinses. Most metals that can be electrodeposited can be adherently plated directly on properly activated beryllium.<sup>(8)</sup> An alternative process involves depositing a chemical displacement zinc film (approximately 0.005 mil in thickness) on the beryllium surface prior to electroplating with copper and/or other metals.<sup>(8)</sup>

Since the majority of applications for beryllium have involved temperatures up to 1500 F, the metallurgy of the coated, composite system is important in the design of beryllium parts. Interdiffusion of a multi-metal system will result in continually changing interfacial layers. Thus, the ultimate operational characteristics of the composite system will be affected by the properties of the various alloys that are formed.

Nickel, iron, chromium, and silver offer promise as preferred metallic coatings on beryllium for elevated-temperature applications.<sup>(5,8)</sup> Nickel-coated beryllium shows no undesirable interfacial alloying after 30 days at 600 F. However, the diffusor between the nickel and beryllium that occurs in 30 days at 932 F results in brittle, low-strength, interfacial alloy layers.<sup>(8)</sup> Iron-coated beryllium shows alloying characteristics similar to that of nickel-coated beryllium but with slightly lower rates of diffusion.<sup>(8)</sup>

Chromium- or silver-coated beryllium, because of lower diffusion rates, offers good chance for electroplated metal coatings on beryllium to be used at high temperatures, above 900 F. However, electroplating techniques for applying these metals on beryllium for protection against oxidation at 1300-1400 F have not been developed yet. Four metallic and non-metallic coatings which were appraised for the protection of beryllium at 1325 F in an argon-plus-air atmosphere (1 vol % air) gave the following life in hours:<sup>(16)</sup>

- Diffused chromium (chromized) coating - CK at 405 hours, failed by oxidation by 580 hours
- Chromized plus 8-mil electroplated nickel coating - nickel separated from chromized coating within 405 hours
- AI\* anodized coatings - failed within 405 hours
- BBC\*\* anodized coatings - CK at 840 hours

Several pretreatment cycles have been developed for adherent electroplating on beryllium. Procedures published by Kolodney in 1952,<sup>(6)</sup> by Beach and Faust in 1953,<sup>(8)</sup> and by Missel in 1960<sup>(11)</sup> constitute the major published plating technology for beryllium. Specific processing details<sup>(8)</sup> and modifications<sup>(4)</sup> follow.

Precleaning for Activation and Plating

Greases and oils are best removed by organic-solvent degreasing (vapor or contact). Residual

\* Atomic International proprietary process.  
 \*\* The Brush Beryllium Company proprietary process.



dirt is removed by cathodic cleaning in alkaline detergent solutions. Proprietary cleaning solutions containing caustic, carbonate, and/or wetting agent, such as those devised for aluminum, copper, magnesium, etc., can be used satisfactorily.

#### Zinc-Immersion Method<sup>(8)</sup>

After precleaning and water rinsing, the clean beryllium part is immersed in the following solution at 185 ± 5 F for about 5 minutes to produce the thin, uniform zinc displacement film on the surface:

Sodium tetrophosphosphate -  $\text{Na}_4\text{P}_2\text{O}_7$  - 120 g/l  
 Zinc sulfate -  $\text{ZnSO}_4 \cdot 7\text{H}_2\text{O}$  - 40 g/l  
 Sodium fluoride - NaF - 7.5 g/l  
 Potassium carbonate -  $\text{K}_2\text{CO}_3$  - 5 g/l  
 pH 7.5-8.0 with sulfuric and/or phosphoric acid.

Other zinc solutions have been used with reported success.<sup>(10,12)</sup>

The adherent zinc film is a basis for subsequent electroplating with copper and/or other metals from solutions designed for plating on zinc. Examples of satisfactory plating baths and conditions for electroplating over zinc immersion-coated beryllium, after water rinsing, are:

#### Copper

Sodium cyanide - NaCN 46 g/l  
 Copper cyanide - CuCN 26 g/l  
 Potassium carbonate -  $\text{K}_2\text{CO}_3$  15 g/l  
 Potassium hydroxide - KOH 7.5 g/l  
 Sodium fluoride - NaF 22.5 g/l  
 pH 13.2 ± 0.1  
 Temperature 130 F

Current density: 30 amp/sq ft for 1 min, followed by 15 amp/sq ft thereafter

#### Iron

Ferrous sulfate -  $\text{FeSO}_4 \cdot 7\text{H}_2\text{O}$  300 g/l  
 Ferrous chloride -  $\text{FeCl}_2 \cdot 4\text{H}_2\text{O}$  42 g/l  
 Ammonium sulfate -  $(\text{NH}_4)_2\text{SO}_4$  15 g/l  
 Boric acid -  $\text{H}_3\text{BO}_3$  35 g/l  
 Sodium formate -  $\text{Na}_2\text{COOH}$  15 g/l  
 Duponol ME (Du Pont) 1 g/l  
 pH 4.0 ± 1  
 Temperature 140 F  
 Current density 40 amp/sq ft

#### Nickel

Nickel sulfate -  $\text{NiSO}_4 \cdot 7\text{H}_2\text{O}$  143 g/l  
 Magnesium sulfate -  $\text{MgSO}_4 \cdot 7\text{H}_2\text{O}$  75 g/l  
 Ammonium chloride -  $\text{NH}_4\text{Cl}$  15 g/l  
 Boric acid -  $\text{H}_3\text{BO}_3$  15 g/l  
 XXXD (Harshaw Chemical Co.) 20 ml/l  
 pH 5.5 ± 0.1  
 Temperature 90 F  
 Current density 15 amp/sq ft

#### Zinc

Zinc sulfate -  $\text{ZnSO}_4 \cdot 7\text{H}_2\text{O}$  240 g/l  
 Ammonium chloride -  $\text{NH}_4\text{Cl}$  15 g/l  
 Aluminum sulfate -  $\text{Al}_2(\text{SO}_4)_3 \cdot \text{H}_2\text{O}$  30 g/l  
 Magnesium sulfate -  $\text{MgSO}_4 \cdot 7\text{H}_2\text{O}$  70 g/l  
 Licorice 1 g/l  
 pH 4.0  
 Temperature 80 ± 5 F  
 Current density 25 amp/sq ft

#### Direct Plating Method 1<sup>(8)</sup>

After being precleaned and water rinsed, the beryllium part is treated as follows:

- (a) Anodic pickle  
 Phosphoric acid (85%  $\text{H}_3\text{PO}_4$ ), 10% (by volume)  
 Hydrochloric acid (38% HCl), 2% (by volume)  
 Temperature, 80 ± 10 F  
 Current density, 15 ± 5 amp/sq ft  
 Time, 1 min
- (b) Chemical pickle (without rinsing)  
 Concentrated nitric acid (70%  $\text{HNO}_3$ )  
 Temperature, 80 ± 10 F  
 Time, 2 min
- (c) Water rinse
- (d) Acid dip  
 Ammonium sulfate -  $\text{NH}_4(\text{SO}_4)_2$ , 100 g/l  
 Sulfuric acid -  $\text{H}_2\text{SO}_4$ , 10 g/l  
 Temperature, 80 ± 10 F  
 Time, 1/2 to 1 min
- (e) Water rinse
- (f) Electroplate

Examples of satisfactory baths and conditions for electroplating directly on activated beryllium are:

#### Nickel and Iron

(See previous section)

#### Aluminum

Aluminum can be plated directly on pretreated beryllium from nonaqueous organic electrolytes.

#### Chromium

Chromic acid solutions passivate beryllium surfaces. Therefore, chromium is plated over another intermediate metal such as copper on beryllium.

#### Silver Strike Bath

Silver cyanide - AgCN 4.5 g/l  
 Sodium cyanide - NaCN 70 g/l  
 Temperature 80 ± 10 F  
 Current density 7.5 amp/sq ft  
 Time 5 minutes

#### Silver Plating

Silver cyanide - AgCN 75 g/l  
 Potassium cyanide - KCN 112 g/l  
 Potassium carbonate -  $\text{K}_2\text{CO}_3$  22.5 g/l  
 pH 13.0 with KOH  
 Temperature 120 F  
 Current density 25 amp/sq ft

#### Tin

Sodium stannate -  $\text{Na}_2\text{SnO}_3 \cdot 3\text{H}_2\text{O}$  150 g/l  
 Sodium hydroxide - NaOH 15 g/l  
 Sodium acetate -  $\text{NaC}_2\text{H}_3\text{O}_2$  22.5 g/l  
 Temperature 1° F  
 Current density 25 amp/sq ft

#### Copper

Sodium cyanide - NaCN 30 g/l  
 Copper cyanide - CuCN 22.5 g/l  
 Sodium carbonate -  $\text{Na}_2\text{CO}_3$  15 g/l  
 Sodium sulfite -  $\text{Na}_2\text{S}_2\text{O}_3$  0.5 g/l  
 Temperature 120 F  
 pH 9 with tartaric acid  
 Current density 25 amp/sq ft

Direct Plating Method II (11)

Nickel coatings (7 mils in thickness) on beryllium have withstood a solar-furnace test in which the underlying beryllium could be heated to its melting point (2341 F) in about 9 seconds. The following processing sequence was reported for adherent nickel plating on beryllium(11)

- (a) Abrasive clean  
Wet 400-grit emory paper
- (b) Water rinse
- (c) Anodic clean  
5 minutes with current density of 50 amp/sq ft in a mild brass-type cleaner at 130 F
- (d) Water rinse
- (e) Acid etch  
Nitric acid (70% HNO<sub>3</sub>) - 5 vol %  
Hydrofluoric acid (48% HF) - 1 vol %  
Water - 94 vol % at room temperature  
30-second immersion
- (f) Water rinse
- (g) Acid activate  
Sulfuric acid - 3.6N H<sub>2</sub>SO<sub>4</sub> at room temperature;  
30- to 60-second immersion
- (h) Water rinse
- (i) Nickel plate  
Proprietary all-sulfamate bath  
pH 3.0 to 3.5;  
Temperature, 130 F;  
Current density, 50 amp/sq ft
- (j) Water rinse and dry

Electroless Coatings

Electroless Nickel

Electroless nickel on beryllium is enjoying considerable popularity as an extremely hard and wear-resistant coating.(4,13) The coating, a nickel-phosphorus alloy containing 6 to 8 per cent phosphorus in solid solution, is nonmagnetic until heated to 750 F and above. The alloy has a coefficient of thermal expansion of  $7.2 \times 10^{-6}$  in./in./F, which compares favorably with that of beryllium,  $6.3 \times 10^{-6}$  in./in./F.

Camera mirrors are electroless-nickel coated, polished, and flash-aluminum coated for aerospace needs. The "Kanigen" electroless-nickel coating polishes like glass, and the thin, vacuum-deposited aluminum provides the needed tarnish resistance.(17) Gyro components were one of the earliest uses of electroless-nickel-coated beryllium. The coating aided wetting of the beryllium component by "potting" compounds.(17)

Pretreatment of beryllium for electroless-nickel plating involves precleaning and application of an immersion zinc film.(18) Kanigen nickel plating is accomplished without any intermediate coating.(17) Reported data on electroless-nickel-coated beryllium are very limited.

Electroless Platinum

Platinum-black coatings on beryllium were investigated by Missel and Greear to consistently

provide a reliable high, total, infrared emittance surface (>0.8) for large parts.(19) The following processing steps were developed and adapted to the coating of hemispherical parts of beryllium using spray-coating techniques:

- (a) Precleaning
  - 1. Abrade with wet emory paper or cloth of 180 mesh or finer
  - 2. Alkaline clean and water rinse
- (b) Activation
  - 1. Acid treat with 3.6N H<sub>2</sub>SO<sub>4</sub> for 2 to 3 minutes and water rinse
  - 2. Zincate treat with zinc chloride solution for 1/2 to 1 minute and water rinse
- (c) Black platinizing
  - 1. Treat with chloroplatinic acid solution for about 2 minutes and water rinse
  - 2. Alcohol rinse and dry.

All solutions are at ambient temperature. Drying after Steps (a2), (b1), or (b2) apparently is not detrimental. The zincate solution (b2) contains 100 g/l of ZnCl<sub>2</sub> in water adjusted to a pH of 4.9 ± 0.1 with acetic acid. The platinizing solution contains 10 g/l of chloroplatinic acid in water.

Anodized Coatings

Oxide coatings have been studied rather extensively for protecting beryllium and for providing control of the thermal-radiation properties for use in spacecraft.(16,20-26)

Air oxidation of beryllium at 900 to 1500 F produces normal hexagonal BeO platelets, parallel to the (0,001) plane, randomly disposed and several hundred angstroms in diameter.(20) Such oxide coatings are not protective. Anodically produced oxide coatings (nitric-chromic acid solutions), also BeO, increased in thickness linearly with the applied voltage.(20) The coating as deposited is not a dielectric. Later studies showed that the anodized layers are crystalline and grow as platelets with a mean diameter of ~60 Å and a mean thickness of ~20 Å.(21)

Chromic acid anodizing of beryllium produces an adherent, glossy-black film of BeO about 0.1 mil in thickness.(22,23) Such coatings were proposed for corrosion protection, for a paint base, and for a heat-radiative surface.(23)

Anodized beryllium was investigated to prevent or retard the interaction of beryllium with uranium dioxide and corrosion by moist CO<sub>2</sub> at temperatures above 1200 F.(24) Better results were observed with chromic acid anodizing than with nitric-chromic acid anodizing. Reaction with UO<sub>2</sub> was avoided. Corrosion in moist CO<sub>2</sub> (3-4 vol % H<sub>2</sub>O) at 1200 F was reduced, but not consistently.

The reflectance of anodized beryllium (1% Cu) is relatively low at the short wave lengths and high at the longer wave lengths; thus, anodized beryllium is attractive for use as a solar heat-collector surface.(25) Sodium hydroxide anodizing in these studies showed more promise than chromic acid anodizing of the beryllium-1% copper alloy.

A proprietary anodizing process for beryllium has been developed and has been shown superior to other anodizing processes for certain needs.(26) This BBC anodizing process was demonstrated capable

for adequate coating and for protecting the intricate neutron-reflector shapes of beryllium that are of interest for the SNAP-8 program.

REFERENCES

- (1) Olofson, C. T., "The Machining of Beryllium", DMIC Memorandum 21, Battelle Memorial Institute, Columbus, Ohio (June 5, 1959).
- (2) Hodge, W., "Some Notes on Safe Handling Practices for Beryllium", DMIC Memorandum 2, Battelle Memorial Institute, Columbus, Ohio (September 22, 1958).
- (3) Stonehouse, A. J., and Beaver, W. W., "Corrosion and Protection of Beryllium Metal", BEC-TR-335, The Brush Beryllium Company, Cleveland, Ohio, December, 1964, paper presented at the Aerospace Metals Symposium, Chicago, Illinois, March 9-13, 1964.
- (4) Beaver, W. W., and Stonehouse, A. J., "Beryllium, Surface Treatments and Coatings", BEC-TR-329, The Brush Beryllium Company, Cleveland, Ohio, August, 1963, paper presented at "A Short Course on Beryllium, Its Properties and Application", University of California, Los Angeles, California, March 4-8, 1963.
- (5) "Beryllium Research and Development Program", coordinated by S. H. Gelles, Technical Documentary Report No. ASD-TDR-62-509, Volume I, Nuclear Metals, Inc., Concord, Massachusetts, Section 6 - "Surface Damage in Beryllium" (M. I. Jacobson, F. M. Almeter, and E. C. Burke of Lockheed Missiles & Space Company, Sunnyvale, California), pp 138-186, Contract No. AF 33(616)-7065 (October, 1962).
- (6) Kolodney, M., "Electroplating on Beryllium", U. S. AEC Report No. AEC-D-2845, University of California, Los Alamos Scientific Laboratory, Los Alamos, New Mexico (August 25, 1950).
- (7) Kolodney, M., "Pretreat of Beryllium Prior to Coating", U. S. Patent 2,588,734 (March 11, 1952).
- (8) Beach, J. G., and Faust, C. L., "Electroplating on Beryllium", J. Electrochem Soc., 100 (6), pp 276-279 (June, 1953).
- (9) Beach, J. G., and Faust, C. L., "Electroplating on Beryllium", U. S. Patent 2,729,601 (January 3, 1956).
- (10) Utz, J. J., "Electroplating on Beryllium and Beryllium Base Alloys", U. S. Patent 2,798,036 (July 2, 1957).
- (11) Missel, L., "Thermal Shock Resistant Plating on Beryllium", Metal Finishing, 58 (3), pp 53-57 (March, 1960).
- (12) Townsend, R. G., "Coating Method", U. S. Patent 2,901,408 (August 25, 1959).
- (13) O'Boyle, D., "A Guide to Specifying Surface Coatings for Beryllium Parts", Machine Design, 33 (26), pp 147-150 (December 21, 1961).
- (14) Gex, R. C., "Electroplating on Magnesium and Beryllium--An Annotated Bibliography", Report No. 58-61-4, Lockheed Aircraft Corporation, Lockheed Missiles and Space Division, Sunnyvale, California (February, 1961).
- (15) Steele, J. R., "General Problems in the Corrosion of Beryllium", The Beryllium Corporation, Reading, Pennsylvania, paper presented at the National Association of Corrosion Engineers, 13th Annual Northeast Regional Conference, New York, New York, October 30-November 2, 1961.
- (16) "High-Temperature Beryllium Corrosion Protection--Phase Report, Coating Development and Evaluation Phase", Report No. ARF-B229-17, Armour Research Foundation now Illinois Institute of Technology, Chicago, Illinois (March 20, 1963).
- (17) Personal communication, Wm. H. Crehan, General Manager, Kanigan Division, General American Transportation Corporation (May, 1964).
- (18) Enthone, Inc., Euplate Plating Cycle No. 18A--for Electroless Nickel Plating on Beryllium (March 22, 1963).
- (19) Missel, L., and Greear, G. R., "Platinum Black Coating of Beryllium", Metal Finishing, 61 (8), pp 46-48,54 (August, 1963).
- (20) Kerr, I. S., and Wilman, H., "The Structure and Growth of Oxide Layers Formed on Beryllium", J. Institute of Metals (London), 84, pp 379-385 (1955).
- (21) Levin, M. L., "The Formation of Crystalline Anodic Oxide Films on Beryllium", Trans. Faraday Society, 54, pp 935-940 (1958).
- (22) Whitby, L., Gowen, E., and Levy, D. J., "Chromic Acid Anodizing of Beryllium Process Parameter", Technical Proceedings, American Electroplaters' Society, pp 106-108 (1961).
- (23) Missel, L., "Chromic Acid Anodizing of Beryllium--Process Development", Technical Proceedings, American Electroplaters' Society, pp 109-111 (1961).
- (24) Vachon, L. J., "Protection of Beryllium Metal by Anodic Films", J. Nuclear Materials, 6 (1), pp 139-141 (1962).
- (25) Janssen, J., Luck, J., and Torborg, R., "Reflectance of Anodized Titanium and Beryllium", Electrochem Technology, 1 (11-12), pp 368-376 (November-December, 1963).
- (26) Nakae, T. S., Ervin, G., Jr., Rausch, J. J., and Stonehouse, A. J., "Anodized Films as Oxidation Protection for Beryllium Metal", NAA, Atomic International, Canoga Park, California; IIT, Armour Research Foundation, Chicago, Illinois; and The Brush Beryllium Company, Cleveland, Ohio, paper presented at the 12th Annual Atomic Energy Commission Corrosion Symposium, Pleasanton, California, May 20-22, 1963.

LIST OF DMIC MEMORANDA ISSUED  
DEFENSE METALS INFORMATION CENTER  
 Battelle Memorial Institute  
 Columbus, Ohio 43201

Copies of the technical memoranda listed below may be obtained from DMIC at no cost by Government agencies and by Government contractors, subcontractors, and their suppliers. Others may obtain copies from the Office of Technical Services, Department of Commerce, Washington, D. C. 20230 (See PB or AD numbers and prices in parentheses.)

Number	Title
1	Thermal Properties of Titanium and Titanium Alloys, August 25, 1958 (PB 161152, \$0.50)
2	Some Notes on Safe Handling Practices for Beryllium, September 22, 1958 (PB 161153, \$0.50)
3	Recent Advances in Titanium Technology, October 24, 1958 (PB 161154, \$0.50)
*4	Effects of High Strain Rates and Rapid Heating on the Tensile Properties of Titanium Alloys, December 29, 1958 (PB 161155, \$0.50)
*5	The Influence of Sheet Thickness on Tensile Properties of Metal Sheet, January 23, 1959 (PB 161156, \$0.50)
6	The Status of Chromium-Base Alloy Development, January 30, 1959 (PB 161157, \$0.50)
7	Implications of Rhenium Research in the Design of Refractory Metals, February 2, 1959 (PB 161158, \$0.50)
8	Elevated-Temperature Mechanical Properties and Oxidation Resistance of Columbium and Its Alloys, February 4, 1959 (PB 161159, \$0.50)
9	Preparation and Analysis of Titanium-Hydrogen Standard Samples, February 9, 1959 (PB 161160, \$0.50)
10	Commercial and Semicommercial Titanium Mill Products, February, 1959
11	Belt Grinding of Titanium Sheet and Plate, March 15, 1959 (PB 161161, \$0.50)
12	Some Metallurgical Considerations in Forging Molybdenum, Titanium, and Zirconium, March 25, 1959 (PB 161162, \$0.50)
*13	Joining of Beryllium, March 30, 1959 (PB 161163, \$0.50)
*14	Physical and Mechanical Properties of Molybdenum and the Mo-0.5Ti Alloy, April 10, 1959 (PB 161164, \$0.50)
15	Mechanical- and Physical-Property Data on Modified 12 Per Cent Chromium Martensitic Stainless Sheet Steels for Airframe Applications, April 18, 1959 (PB 161165, \$0.50)
16	Glass-Bonded Refractory Coatings for Iron- or Nickel-Base Alloys, April 25, 1959 (PB 161166, \$0.50)
17	Future Application Trends for Titanium and Steel in Military Aircraft, May 8, 1959 (PB 161167, \$0.50)
18	Fabrication of 17-7PH and PH15-7Mo Stainless Steel by Bend Rolling, Deep Drawing, and Spinning, May 15, 1959 (PB 161168, \$0.50)
19	The Availability and Properties of Rhenium, May 22, 1959 (PB 161169, \$0.50)
20	The Properties of Magnesium-Thorium Alloys, May 29, 1959 (PB 161170, \$0.50)
21	Machining of Beryllium, June 5, 1959 (PB 161171, \$0.50)
*22	Routing of Titanium Sheet, June 12, 1959 (PB 161172, \$0.50)
23	Band Sawing of Titanium and Titanium Alloys, July 1, 1959 (PB 161173, \$0.50)
24	Hacksawing of Titanium and Titanium Alloys, July 6, 1959 (PB 161174, \$0.50)
25	Profile Milling Titanium and Its Alloys, July 10, 1959 (PB 161175, \$0.50)
26	Spindle Shaping of Titanium Sheet, July 15, 1959 (PB 161176, \$0.50)
*27	Arc Welding of High-Strength Steels for Aircraft and Missile Structures, July 31, 1959 (PB 161177, \$0.50)
28	Review of Electrical Machining Methods, August 5, 1959 (PB 161178, \$0.50)
*29	Nitriding of Titanium, August 12, 1959 (PB 161179, \$0.50)
30	Milling of High-Strength Steels in the Hardness Range of 330 to 560 Brinell, August 17, 1959 (PB 161180, \$0.50)
31	Drilling High-Strength Steels Heat Treated to 330 to 560 Brinell Hardness, August 24, 1959 (PB 161181, \$0.50)
32	Physical and Mechanical Properties of Tantalum, August 28, 1959 (PB 161182, \$0.50)
33	Titanium Fabrication and Reliability Problems in Aircraft, September 4, 1959 (PB 161183, \$0.50)
*34	Fabrication of Pure Columbium, September 11, 1959 (PB 161184, \$0.50)
35	Procedures for Electroplating Coatings on Refractory Metals, October 9, 1959 (PB 161185, \$0.50)
36	Heat Capacity of Beryllium, October 19, 1959 (PB 161186, \$0.50)
37	Procedures for the Metallographic Preparation of Beryllium, Titanium, and Refractory Metals, October 26, 1959 (PB 161187, \$0.50)
38	The Welding of Wrought Age-Hardenable Nickel-Base Alloys for Service at Elevated Temperatures, November 25, 1959 (PB 161188, \$0.50)
*39	Development of High-Strength Steels by Working of Metastable Austenite, November 30, 1959 (PB 161189, \$0.50)
40	A Brief Review of Refractory Metals, December 3, 1959 (PB 161190, \$0.50)
41	The Properties of Boron, January 4, 1960 (PB 161191, \$0.50)
42R	Standard Designations of Alloys for Aircraft and Missiles, May 24, 1961 (AD 233728, \$0.50) Obsolete, Memo 177
43	Notes on Mechanical Testing Techniques at Very Low Temperatures, February 19, 1960 (PB 161193, \$0.50)
44	Refractory Materials, February 26, 1960 (PB 161194, \$0.50)
45	Recent Developments in Titanium Brazing, March 4, 1960 (PB 161195, \$0.50)
46	Fatigue Data on Precipitation-Hardenable Stainless Steel, March 11, 1960 (PB 161196, \$0.50)
47	Selected References on Making High-Temperature Alloys by Powder Metallurgy, March 18, 1960 (PB 161197, \$0.50)
48	Brazing for High-Temperature Service, March 29, 1960 (PB 161198, \$0.50)
49	The Determination of Oxygen, Nitrogen, Hydrogen, and Carbon in Molybdenum, Tungsten, Columbium, and Tantalum, March 31, 1960 (PB 161199, \$0.50)
*50	Diffusion Rates and Solubilities of Interstitials in Refractory Metals, April 4, 1960 (PB 161200, \$0.50)
51	Bibliography on Explosive Metal Working, April 7, 1960 (PB 161201, \$0.50)
52	Review of Problems in Using Flat-Rolled Materials in Air- and Space-Weapon Systems, April 14, 1960 (PB 161202, \$0.50)
53	Notes on the Diffusion Bonding of Metals, April 20, 1960 (PB 161203, \$0.50)
54	Problems with Restraint in Heavy Weldments, April 29, 1960 (PB 161204, \$0.50)

\* DMIC supply exhausted; copies may be ordered from OTS.

## LIST OF DMIC MEMORANDA ISSUED (CONTINUED)

Number	Title
*55	Selected References on Brittle Fracture, May 5, 1960 (PB 161205, \$0.50)
*56	Welded Fabrication of Steel Solid-Propellant Rocket-Motor Cases, May 31, 1960 (PB 161206, \$0.50)
*57	References to Research on High-Emissivity Surfaces, June 27, 1960 (PB 161207, \$0.50)
58	Turning of High-Strength Steels in the Hardness Range of 330 to 560 Brinell, July 15, 1960 (PB 161208, \$0.50)
*59	Metallurgical Characteristics of A-286 Alloy, July 26, 1960 (PB 161209, \$0.50)
*60	Stress-Corrosion Cracking of Ti-5Al-2.5Sn; August 4, 1960 (PB 161210, \$0.50)
61	Selected References to ASTIA Documents on Machining, August 11, 1960 (PB 161211, \$0.50)
62	Effects of Rate of Heating to Aging Temperature on Tensile Properties of Ti-2.5Al-16V Alloys, August 18, 1960 (PB 161212, \$0.50)
*63	Notes on Large-Size Electrical Furnaces for Heat Treating Metal Assemblies, August 25, 1960 (PB 161213, \$0.50)
64	Recent Developments in Superalloys, September 8, 1960 (PB 161214, \$0.50)
65	Compatibility of Rocket Propellants with Materials of Construction, September 15, 1960 (PB 161215, \$0.50)
*66	Physical and Mechanical Properties of the Cobalt-Chromium-Tungsten Alloy WI-52, September 22, 1960 (PB 161216, \$0.50)
67	Development of Refractory Metal Sheet in the United States, September 20, 1960 (PB 161217, \$0.50)
68	Some Physical Properties of Martensitic Stainless Steels, September 28, 1960 (PB 161218, \$0.50)
69	Welding of Columbium and Columbium Alloys, October 24, 1960 (PB 161219, \$0.50)
70	High Velocity Metalworking Processes Based on the Sudden Release of Electrical Energy, October 27, 1960 (PB 161220, \$0.50)
71	Explosive Metalworking, November 3, 1960 (PB 161221, \$0.50)
72	Emissivity and Emittance--What are They?, November 10, 1960 (PB 161222, \$0.50)
73	Current Nickel-Base High-Temperature Alloys, November 17, 1960 (PB 161223, \$0.50)
74	Joining of Tungsten, November 24, 1960 (PB 161224, \$0.50)
75	Review of Some Unconventional Methods of Machining, November 29, 1960 (PB 161225, \$0.50)
76	Production and Availability of Some High-Purity Metals, December 2, 1960 (PB 161226, \$0.50)
*77	Rocket Nozzle Testing and Evaluation, December 7, 1960 (PB 161227, \$0.50)
78	Methods of Measuring Emittance, December 27, 1960 (PB 161228, \$0.50)
*79	Preliminary Design Information on Recrystallized Mo-0.5Ti Alloy for Aircraft and Missiles, January 16, 1961 (PB 161229, \$0.50)
80	Physical and Mechanical Properties of Some High-Strength Fine Wires, January 20, 1961 (PB 161230, \$0.50)
81	Design Properties as Affected by Cryogenic Temperatures (Ti-6Al-4V, AISI 4340, and 7079-T6 Alloys), January 24, 1961 (PB 161231, \$0.50)
82	Review of Developments in Iron-Aluminum-Base Alloys, January 30, 1961 (PB 161232, \$0.50)
*83	Refractory Metals in Europe, February 1, 1961 (PB 161233, \$0.50)
84	The Evolution of Nickel-Base Precipitation-Hardening Superalloys, February 6, 1961 (PB 161234, \$0.50)
85	Pickling and Descaling of High-Strength, High-Temperature Metals and Alloys, February 8, 1961 (PB 161235, \$0.50)
86	Superalloy Forgings, February 10, 1961 (PB 161236, \$0.50)
87	A Statistical Summary of Mechanical-Property Data for Titanium Alloys, February 14, 1961 (PB 161237, \$0.50)
88	Zinc Coatings for Protection of Columbium from Oxidation at Elevated Temperatures, March 3, 1961 (PB 161238, \$0.50)
89	Summary of Present Information on Impact Sensitivity of Titanium When Exposed to Various Oxidizers, March 5, 1961 (PB 161239, \$0.50)
90	A Review of the Effects of Starting Material on the Processing and Properties of Tungsten, Molybdenum, Columbium, and Tantalum, March 13, 1961 (PB 161240, \$0.50)
91	The Emittance of Titanium and Titanium Alloys, March 17, 1961 (PB 161241, \$0.50)
92	Stress-Rupture Strengths of Selected Alloy, March 23, 1961 (AD 255075, \$0.50)
93	A Review of Recent Developments in Titanium and Titanium Alloy Technology, March 27, 1961 (PB 161243, \$0.50)
*94	Review of Recent Developments in the Evaluation of Special Metal Properties, March 28, 1961 (PB 161244, \$0.50)
*95	Strengthening Mechanisms in Nickel-Base High-Temperature Alloys, April 4, 1961 (PB 161245, \$0.50)
96	Review of Recent Developments in the Technology of Molybdenum and Molybdenum-Base Alloys, April 7, 1961 (PB 161246, \$0.50)
*97	Review of Recent Developments in the Technology of Columbium and Tantalum, April 10, 1961 (PB 161247, \$0.50)
*98	Electropolishing and Chemical Polishing of High-Strength, High-Temperature Metals and Alloys, April 12, 1961 (PB 161248, \$0.50)
*99	Review of Recent Developments in the Technology of High-Strength Stainless Steels, April 14, 1961 (PB 161249, \$0.50)
100	Review of Current Developments in the Metallurgy of High-Strength Steels, April 20, 1961 (PB 161250, \$0.50)
101	Statistical Analysis of Tensile Properties of Heat-Treated Mo-0.5Ti Sheet, April 24, 1961 (AD 255456, \$0.50)
*102	Review of Recent Developments on Oxidation-Resistant Coatings for Refractory Metals, April 26, 1961 (AD 255278, \$0.50)
*103	The Emittance of Coated Materials Suitable for Elevated-Temperature Use, May 4, 1961 (AD 256479, \$2.75)
*104	Review of Recent Developments in the Technology of Nickel-Base and Cobalt-Base Alloys, May 5, 1961 (AD 255659, \$0.50)
*105	Review of Recent Developments in the Metallurgy of Beryllium, May 10, 1961 (AD 256206, \$0.50)
106	Survey of Materials for High-Temperature Bearing and Sliding Applications, May 12, 1961 (AD 257408, \$2.00)
107	A Comparison of the Brittle Behavior of Metallic and Nonmetallic Materials, May 16, 1961 (AD 258042, \$0.50)
108	Review of Recent Developments in the Technology of Tungsten, May 18, 1961 (AD 256633, \$0.50)
*109	Review of Recent Developments in Metals Joining, May 25, 1961 (AD 256852, \$0.50)
110	Glass Fiber for Solid-Propellant Rocket-Motor Cases, June 6, 1961 (AD 258862, \$0.75)
111	The Emittance of Stainless Steels, June 12, 1961 (AD 259283, \$0.50)
112	Review of Recent Developments in the Evaluation of Special Metal Properties, June 27, 1961 (AD 259.77, \$0.50)

\* MIC supply exhausted; copies may be ordered from OTS.

LIST OF DMIC MEMORANDA ISSUED (CONTINUED)

<u>Number</u>	<u>Title</u>
*113	A Review of Recent Developments in Titanium and Titanium Alloy Technology, July 3, 1961 (AD 259178, \$0.50)
*114	Review of Recent Developments in the Technology of Molybdenum and Molybdenum-Base Alloys, July 5, 1961 (AD 259449, \$0.50)
115	Review of Recent Developments in the Technology of Columbium and Tantalum, July 7, 1961 (AD 259840, \$0.50)
116	General Recommendations on Design Features for Titanium and Zirconium Production-Melting Furnaces, July 19, 1961 (AD 260099, \$0.50)
117	Review of Recent Developments in the Technology of High-Strength Stainless Steels, July 14, 1961 (AD 259943, \$0.50)
*118	Review of Recent Developments in the Metallurgy of High-Strength Steels, July 21, 1961 (AD 259986, \$0.50)
119	The Emittance of Iron, Nickel, Cobalt and Their Alloys, July 25, 1961 (AD 261336, \$2.25)
*120	Review of Recent Developments on Oxidation-Resistant Coatings for Refractory Metals, July 31, 1961 (AD 261293, \$0.50)
121	Fabricating and Machining Practices for the All-Beta Titanium Alloy, August 3, 1961 (AD 262496, \$0.50)
*122	Review of Recent Developments in the Technology of Nickel-Base and Cobalt-Base Alloys, August 4, 1961 (AD 261292, \$0.50)
*123	Review of Recent Developments in the Technology of Beryllium, August 18, 1961 (AD 262497, \$0.50)
124	Investigation of Delayed-Cracking Phenomenon in Hydrogenated Unalloyed Titanium, August 30, 1961 (AD 263164, \$0.50)
*125	Review of Recent Developments in Metals Joining, September 1, 1961 (AD 262905, \$0.50)
126	A Review of Recent Developments in Titanium and Titanium Alloy Technology, September 15, 1961 (AD 263167, \$0.50)
127	Review of Recent Developments in the Technology of Tungsten, September 22, 1961 (AD 263888, \$0.50)
128	Review of Recent Developments in the Evaluation of Special Metal Properties, September 27, 1961 (AD 263994, \$0.50)
129	Review of Recent Developments in the Technology of Molybdenum and Molybdenum-Base Alloys, October 6, 1961 (AD 264291, \$0.50)
130	Review of Recent Developments in the Technology of Columbium and Tantalum, October 10, 1961 (AD 264983, \$0.50)
131	Review of Recent Developments in the Technology of High-Strength Stainless Steels, October 13, 1961 (AD 264984, \$0.50)
132	Review of Recent Developments in the Metallurgy of High-Strength Steels, October 20, 1961 (AD 265135, \$0.50)
133	Titanium in Aerospace Applications, October 24, 1961 (AD 266927, \$1.50)
134	Machining of Superalloys and Refractory Metals, October 27, 1961 (AD 268081, \$1.00)
135	Review of Recent Developments in the Technology of Nickel-Base and Cobalt-Base Alloys, October 31, 1961 (AD 266004, \$0.50)
136	Fabrication of Tungsten for Solid-Propellant Rocket Nozzles, November 2, 1961 (AD 268311, \$0.75)
137	Review of Recent Developments on Oxidation-Resistant Coatings for Refractory Metals, November 8, 1961 (AD 266469, \$0.50)
138	Review of Recent Developments in the Technology of Beryllium, November 16, 1961 (AD 267079, \$0.50)
*139	Review of Recent Developments in the Technology of Tungsten, November 24, 1961 (AD 268082, \$0.50)
*140	Review of Recent Developments in Metals Joining, December 6, 1961 (AD 268312, \$0.50)
141	The Emittance of Chromium, Columbium, Molybdenum, Tantalum, and Tungsten, December 10, 1961 (AD 269784, \$1.25)
*142	Effects of Moderately High Strain Rates on the Tensile Properties of Metals, December 18, 1961 (AD 270167, \$1.00)
143	Notes on the Forging of Refractory Metals, December 21, 1961 (AD 271030, \$1.50)
144	Review of Recent Developments in Titanium Alloy Technology, December 29, 1961 (AD 269209, \$0.50)
145	The Use of Nickel-Base Alloys in the Rotating Parts of Gas Turbines for Aerospace Applications, January 11, 1962 (AD 271174, \$0.75)
146	Magnesium-Lithium Alloys - A Review of Current Developments, February 6, 1962 (AD 272683, \$0.75)
147	An Evaluation of Materials for Rocket-Motor Cases Based on Minimum-Weight Concepts, March 8, 1962 (AD 273297, \$1.00)
*148	The Emittance of Ceramics and Graphites, March 28, 1962 (AD 274148, \$2.50)
149	Methods of Strengthening the Ultrahigh-Strength Steels, April 2, 1962 (AD 275041, \$1.00)
150	Compilation of Tensile Properties of High-Strength Alloys, April 23, 1962 (AD 275263, \$1.25)
151	Compatibility of Propellants 113 and 114B2 with Aerospace Structural Materials, April 27, 1962 (AD 275427, \$0.50)
*152	Electron-Beam Welding of Tungsten, May 21, 1962 (AD 276283, \$0.50)
153	Brazing and Bonding of Columbium, Molybdenum, Tantalum, Tungsten and Graphite, June 11, 1962 (AD 278193, \$0.75)
154	The Effects of Decarburization on the Properties of Ultrahigh-Strength Steels, June 18, 1962 (AD 278194, \$0.75)
155	The Effects of Solutes on the Ductile-to-Brittle Transition in Refractory Metals, June 28, 1962 (AD 278652, \$1.75)
156	Properties of Mar-Aging Steels, July 2, 1962 (AD 281888, \$1.50)
*157	A Compilation of the Tensile Properties of Tungsten, September 11, 1962 (AD 283572, \$1.00)
158	Summary of Briefings on Refractory Metal Fasteners, October 8, 1962 (AD 287287, \$1.00)
159	Nondestructive Testing of Solid-Propellant Rocket Motors, October 24, 1962 (AD 287803, \$0.50)

\* DMIC supply exhausted; copies may be ordered from OTS.

## LIST OF DMIC MEMORANDA ISSUED (CONTINUED)

Number	Title
*160	Identification of Microconstituents in Superalloys, November 15, 1962 (AD 289664, \$0.75)
161	Electron Microscopic Fractography, December 21, 1962 (AD 295029, \$1.00)
162	Report on Meeting to Review Maraging Steel Projects, December 28, 1962 (AD 296040, \$0.75)
163	Reactivity of Metals with Liquid and Gaseous Oxygen, January 15, 1963 (AD 297124, \$0.75)
164	A Discussion of the Fracture Toughness of Several Stainless Steels in Sheet Form, January 31, 1963 (AD 298204, \$3.60)
165	Review of Uses for Depleted Uranium and Nonenergy Uses for Natural Uranium, February 1, 1963 (AD 299705, \$0.75)
166	Literature Survey on the Effect of Sonic and Ultrasonic Vibrations in Controlling Grain Size During Solidification of Steel Ingots and Weldments, May 15, 1963 (AD 410538, \$1.00)
167	Notes on Large-Size Furnaces for Heat Treating Metal Assemblies, May 24, 1963 (A Revision of DMIC Memorandum 63) (AD 410282, \$0.50)
168	Some Observations on the Arc Melting of Tungsten, May 31, 1963 (AD 409824, \$0.50)
169	Weldability Studies of Three Commercial Columbium-Base Alloys, June 17, 1963 (AD 415263, \$0.75)
170	Creep of Columbium Alloys, June 24, 1963 (AD 424097, \$2.25)
171	A Tabulation of Designations, Properties, and Treatments of Titanium and Titanium Alloys, July 15, 1963 (AD 424412, \$0.50)
172	Production Problems Associated with Coating Refractory Metal Hardware for Aerospace Vehicles, July 26, 1963
173	Reactivity of Titanium with Gaseous $N_2O_4$ Under Conditions of Tensile Rupture, August 1, 1963 (AD 419555, \$0.50)
174	Some Design Aspects of Fracture in Flat Sheet Specimens and Cylindrical Pressure Vessels, August 9, 1963 (AD 420376, \$0.75)
175	Consideration of Steels with Over 150,000 psi Yield Strength for Deep-Submergence Hulls, August 16, 1963 (AD 420873, \$0.50)
176	Preparation and Properties of Fiber-Reinforced Structural Materials, August 22, 1963 (AD 422242, \$0.75)
177	Designations of Alloys for Aircraft and Missiles, September 4, 1963 (AD 424998, \$1.75)
178	Some Observations on the Distribution of Stress in the Vicinity of a Crack in the Center of a Plate, September 18, 1963 (AD 422463, \$1.00)
179	Short-Time Tensile Properties of the Co-20Cr-15W-10Ni Cobalt-Base Alloy, September 27, 1963 (AD 425922, \$2.75)
180	The Problem of Hydrogen in Steel, October 1, 1963 (AD 425124, \$1.00)
181	Report on the Third Maraging Steel Project Review, October 7, 1963 (AD 425125, \$1.50)
182	The Current Status of the Welding of Maraging Steels, October 16, 1963 (AD 425714, \$1.00)
183	The Current Status and 1970 Potential for Selected Defense Metals, October 31, 1963 (AD 425604, \$1.25)
184	A Review and Comparison of Alloys for Future Solid-Propellant Rocket-Motor Cases, November 15, 1963 (AD 430165, \$1.25)
185	Classification of DMIC Reports and Memoranda by Major Subject, January 15, 1964
186	A Review of Some Electron-Microscopic Fractographic Studies of Aluminum Alloys, February 5, 1964 (AD 434212, \$0.50)
187	Some Observations on the Electron-Microscopic Fractography of Embrittled Steels, February 19, 1964 (AD 602288, \$2.25)
188	A Review of Available Information on the Welding of Thick Titanium Plate in the USSR, March 6, 1964
189	A Review of Dimensional Instability in Metals, March 19, 1964
190	Continued Observations on the Distribution of Stress in the Vicinity of a Crack in the Center of a Plate, April 14, 1964
191	Observations on Delayed Cracking in Welded Structures of Unalloyed Titanium Sheet, April 29, 1964
192	Summary of the Eighth Meeting of the Refractory Composites Working Group, April 20, 1964
193	Mechanical and Physical Properties of Three Superalloys--MAR-M200, MAR-M302, and MAR-M322, May 6, 1964
194	Porosity in Titanium Welds, June 1, 1964
195	The Production of Powder-Metallurgy Tungsten Sheet and Plate, July 20, 1964
196	Report on the Fourth Maraging-Steel Project Review, August 19, 1964