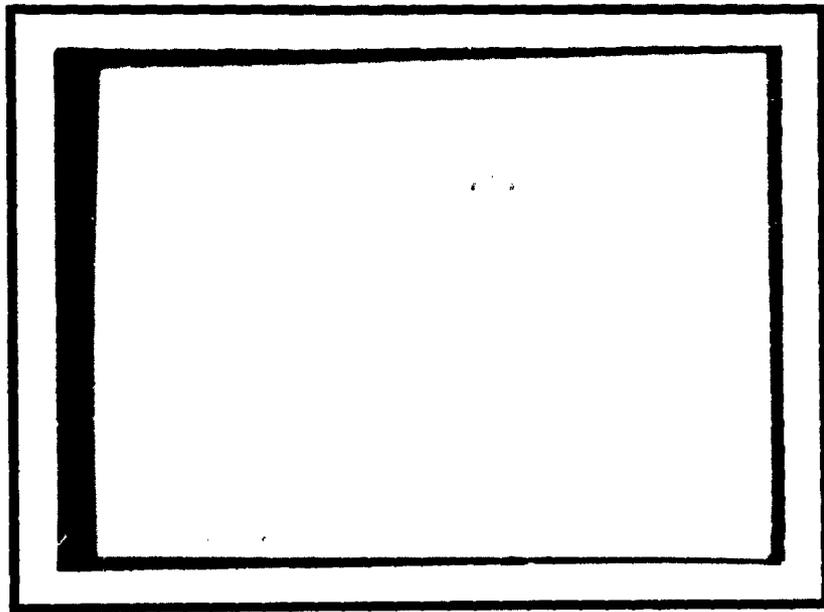


CAT# ASTIA
AS AD NO.

402158



ASTIA
APR 25 1963
LIBRARY
ASTIA

HORIZONS INCORPORATED

2905 EAST 79TH STREET • CLEVELAND 4, OHIO



Interim Report No. 2
Reinforcement of Nickel Chromium
Alloys with Sapphire Whiskers
Prepared under Navy, Bureau of Weapons
Contract N0w 63-0138-c

For: Department of the Navy
Bureau of Naval Weapons
Washington 25, D. C.

From: Horizons Incorporated
2905 East 79th Street
Cleveland 4, Ohio

Date: April 22, 1963

Period: 29 December 1962 - 28 March 1963

Written by: Robert H. Kelson



Reproduction in whole or in part is permitted for any purpose of the United States Government.

"Qualified requestors may obtain copies of this report directly from ASTIA."

ABSTRACT

It has been shown that oxygen is required for the formation of a bond between iron and alumina whiskers. This oxygen may be introduced as naturally-occurring oxides on the surfaces of iron powder. Reinforcement has again been demonstrated in iron-alumina whisker composites subjected to short heating cycles. Evidence has been obtained that long heating cycles and excessive oxide content result in destructive attack on the alumina whiskers. Methods are projected for introducing the necessary oxygen by pretreating the whiskers.

TABLE OF CONTENTS

| <u>Section</u> | | <u>Page</u> |
|----------------|-------------------|-------------|
| I | Introduction..... | 1 |
| II | Discussion..... | 12 |
| | Distribution List | |

LIST OF TABLES

| <u>Table</u> | | <u>Page</u> |
|--------------|---|-------------|
| I | Effect of Oxygen on Retention of Alumina Whiskers in Iron..... | 5 |
| II | Effect of Oxygen as CO ₂ on Retention of Alumina Whiskers in Iron..... | 6 |
| III | Effect of Time and Temperature on Interaction of Iron Powder and Al ₂ O ₃ Fibers..... | 8 |
| IV | Effect of Low Oxide Content on Wetting of Al ₂ O ₃ by Iron..... | 9 |
| V | Effect of Time on Bonding of Iron and Alumina Whiskers..... | 11 |

I. Introduction

This report describes research directed toward the development of high-strength composites of refractory metals and alumina whiskers. The work is sponsored by the Department of the Navy, Bureau of Naval Weapons.

A limited reinforcement of 80/20 nickel-chromium alloy and of pure iron by alumina (corundum) whiskers has been shown. It was found that an increase of 34% in the tensile strength of pure hydrogen-reduced iron could be achieved by the addition of 8 weight percent of alumina whiskers; 80/20 Ni/Cr showed an increase of 28% with 3.8 weight percent whiskers.

The purpose of the research program has been to obtain a detailed understanding of the process by which these metallurgical bonds were formed, in order to extend the reinforcement to higher levels and to a wider range of metals.

In the initial phase of the program, powdered metals, iron, 317 stainless steel and 80/20 Ni/Cr were melted on pure sintered alumina disks, to observe adhesion. It was found that the trivalent oxides tended to inhibit the formation of the necessary strong metallurgical bond. On the other hand, no bond could be formed without a small amount of oxygen present. To this end, iron was selected for intensive study.

The amount of oxygen needed to form a layer of iron spinel over the entire surface of the fiber was calculated to be 3.47×10^{-4} gram per gram fiber. If this amount of oxygen can be made available at the fiber surface, formation of the bond is possible. There are, however, three conditions defined

by experiment to date which must be met. These are:

- (a) The fiber must be completely surrounded by metal, in contact with its surface, in the molten state.
- (b) Oxygen must be made available at this interface.
- (c) The resulting structure must be compatible with both iron and fiber, without destructive attack on the fiber itself.

Condition (a) means that the metal must in effect "wet" the fiber; and further, the fiber must be kept from floating out of the metal until the bond is formed.

Condition (b) may be met by the presence of dissolved oxygen in the metal, by the presence of metal or other oxides, or by transport through the molten surface of the metal.

Condition (c) implies a severely limited reaction of the fiber surface with oxygen and metal.

The present report is concerned with a series of four crucial experiments based on these hypotheses:

1. No significant interaction of iron and alumina whiskers can occur in the complete absence of oxygen.
2. Profound interaction between iron and whiskers results when the naturally-occurring iron oxides are present.
3. The interaction of iron oxides and alumina whiskers is temperature dependent.
4. The interaction of iron and alumina whiskers is time dependent.

To verify hypothesis (1) above, a layer of alumina whiskers was placed in a zircon boat, on top of a layer of iron powder. The boat was put in a tube

furnace, with an atmosphere of hydrogen and the temperature raised to 2900°F. After cooling in argon, the sample was visually and microscopically examined. The iron had formed a clean ingot below the fibers. The fibers were unchanged in appearance and there was no evidence of any interaction. The experiment was repeated, with a layer of fiber between the layers of iron powder. The result was the same: all the iron was found on the bottom of the boat, and no evidence of interaction could be found.

Samples were again prepared as above, and brought to melting in dry hydrogen. At a temperature of 2850°F, a 97/3 volume percent mixture of argon and oxygen was then introduced into the furnace and held 10 minutes, after which cooling was done in argon. In the case of fibers placed on top of iron powder, the fibers were not wetted, but had taken on a brownish color. The iron had drawn up into droplets.

When the fiber was placed between two layers of iron, the fibers were deeply discolored, and there was evidence of attack by the iron oxide. The fibers were not included intact within the iron ingots, however, and the tensile strength of the iron (75,000 psi) showed no reinforcement. Notebook reference 881-27-b, c, d, e.

Two subsequent experiments further clarify the role of oxygen in the fiber-iron system. Intimate mixture of alumina whiskers and iron powder was prepared, with 15 weight percent whiskers on the total. These were treated in the following way: a boat containing the mixture and a second boat containing iron powder only were placed in a tube furnace, and the temperature raised to 1800°F in an atmosphere of dry hydrogen. Argon was then substituted for the hydrogen, and the furnace temperature was raised to 2800°F \pm 20°F.

In one experiment, pure oxygen was added to the argon; in a second experiment, CO₂ was added. The results are given in Tables I and II.

Table I indicates the disastrous effect of excessive oxygen: with 10 % the fibers and oxide were reduced to a brownish mass. There was no evidence of fiber retention. At lower oxygen levels, the metal appeared to be attached to the fiber as shot. These were separated magnetically from the loose whiskers and alumina content determined analytically. The high values proved to be illusory, however. The tensiles of the remelted materials showed no increase over the blank.

Table II shows the effect of very small amounts of oxygen, introduced as CO₂. There was little evidence of fiber retention; analytical data support this observation.

These experiments clearly show that without oxygen present, there is no interaction between iron and alumina whiskers. Earlier experiments had suggested this conclusion.

The use of the tube furnace requires that heating times be relatively long. In order to provide shorter cycles, a high-frequency induction furnace was put in operation. Atmospheres are contained within a transparent quartz envelope.

This equipment was used to investigate the infiltration of fiber by iron with its naturally-occurring oxides.

One gram of clean fiber was placed in the bottom of a high-temperature glass tube, and five grams of iron powder containing 6.4 % oxygen was poured on top. This was put in a cylindrical graphite susceptor and set in the induction furnace. After purging with argon, the temperature was rapidly raised



TABLE I
EFFECT OF OXYGEN ON RETENTION OF
ALUMINA WHISKERS IN IRON

| <u>Sample</u> | <u>W/o Fiber</u> | <u>V/o Oxygen in Argon</u> | <u>Result</u> | <u>W/o Al₂O₃ Retained</u> |
|---------------|------------------|--------------------------------|-------------------------------|---|
| 881-24-A | 0 | 1.0 | Good sample | --- |
| 881-24-B | 15 | 1.0 | Fibers black, metal beads. | 17.09 |
| 881-24-C | 0 | 5.0 | Black glaze on metal. | --- |
| 881-24-D | 15 | 5.0 | Fibers black, metal beads. | 11.2 |
| 881-25-A | 0 | 10.0 | Black-brown slag | --- |
| 881-25-B | 15 | 10.0 | Black brown slag | --- |

TABLE II
EFFECT OF OXYGEN AS CO₂ ON RETENTION OF
ALUMINA WHISKERS IN IRON

| <u>Sample</u> | <u>W/o Fiber</u> | <u>V/o CO₂ in Argon</u> | <u>Result</u> | <u>W/o Al₂O₃ Retained</u> |
|---------------|------------------|--|----------------|---|
| 881-17-A | 0 | 0 | No Wetting | --- |
| 881-17-B | 15 | 0 | No Wetting | 0.135 |
| 881-17-C | 15 | 5 | Little Wetting | 0.025 |
| 881-18-C | 15 | 20 | Some Slag | 0.032 |

to 2800°F ± 20°F and held five minutes. There was no infiltration of the fiber by the molten iron. A series of experiments of this type is described in Table III.

The data indicate that, with the oxygen introduced as metal oxide, it is necessary that (a) the temperature be considerably higher than the melting point of the metal, and (b) that sufficient time be allowed for wetting and infiltration to take place. This experiment strongly suggests that the fibers may be destroyed rather than wetted. The appearance would be macroscopically the same. Metallurgical examination of the resulting pieces was not conclusive: intact large whiskers may be seen but the small wool-like filaments could not definitely be identified in the polished and etched surfaces.

A further experiment, using iron which had been heated in hydrogen to reduce the oxide content, is described in Table IV. Again the induction furnace was used. Little or no metal-fiber interaction could be seen. The metal flowed around the fibers, along the tube wall, leaving them essentially intact.

Since the reaction is time-dependent, a series of experiments to explore the effect of very short heating times was then undertaken. For this purpose, a vertical furnace with a hollow cylindrical silicon carbide heating element was used. This furnace may be brought to the desired temperature before the sample is introduced, which permits heating times limited only by the time required for the sample to reach thermal equilibrium.

Ten grams of iron powder containing 6.4 weight percent oxygen was mixed intimately with 0.8 grams of cleaned alumina whiskers. All casual aluminum metal was removed from the whiskers by alkali treatment. The mixture was placed in a dry high temperature glass tube, connected to a controlled pressure of



TABLE III
EFFECT OF TIME AND TEMPERATURE ON INTERACTION
OF IRON POWDER AND Al_2O_3 FIBERS

| <u>Sample</u> | <u>Temperature (°F)</u> | <u>Time (Min.)</u> | <u>Result</u> |
|---------------|-----------------------------|------------------------|--|
| 886-19-2 | 2820 | 5 | Melted ingot on top of fiber. No infiltration. |
| 886-20-1 | 2980 | 0.1 | No infiltration. |
| 886-20-2 | 2980 | 0.5 | Little infiltration. |
| 886-20-3 | 2980 | 1.0 | Little infiltration. |
| 886-20-4 | 2980 | 2.0 | Complete absorption of fiber. |
| 886-19 | 2980 | 5.0 | Complete absorption of fiber. Tensile 80,000 psi. |



TABLE IV
EFFECT OF LOW OXIDE CONTENT ON WETTING OF
 Al_2O_3 BY IRON

| <u>Sample</u> | <u>Fiber Content ($\frac{w}{o}$)</u> | <u>Oxygen Content of Fe ($\frac{w}{o}$)</u> | <u>Temperature ($^{\circ}F$)</u> | <u>Time (Min.)</u> | <u>Result</u> |
|---------------|---|--|---|------------------------|------------------------------|
| 881-32-A | 20 | 0.545 | 3000 | 12 | Metal flowed past fibers. |
| 881-32-B | 20 | 0.545 | 3000 | 12 | Same. |
| 881-32-C | 20 | 1.12 | 3000 | 12 | Same. |



20 mm Hg of argon. This tube was lowered into the vertical tube furnace preheated to 2900°F, and held two minutes. The resulting ingot showed no loose fibers. It was divided in two equal parts: one was held for tensile test; the other was returned to the furnace for an additional two minutes heating.

A third sample was similarly treated, except that a total heating time of six minutes was used. Blanks containing no fiber were treated in a similar way.

The technique for measuring tensile strengths has been improved by the development of a device for grinding an accurately round, smooth section of reduced diameter in the specimen. This method has eliminated clamp breaks entirely and materially improved the consistency of tensile data.

The tensiles on the above samples are given in Table V. In addition to the tensiles, microscope examination showed large fibers in the two minute sample, but no clearly defined small fibers. It is evident that the attack of the oxide on the fiber was minimal at the short time, with only the smallest whiskers destroyed. However, with longer heating times, the fibers were progressively destroyed, and the tensile strength reduced to that of the blank. An exploratory experiment using alumina whiskers which had been coated with nickel showed no beneficial effect from the coating. The coated fibers were by weight 70/30 Ni/Al₂O₃.

Fibers so coated were tightly packed in a mullite tube and heated to 2750°F in argon. The metal separated into small beads or shot, leaving the fibers clean.

In a second experiment, the coated fibers were mixed with Fe powder containing 0.54 % oxygen as surface oxides, and heated to 2900°F in argon.

TABLE V
EFFECT OF TIME ON BONDING OF IRON
AND ALUMINA WHISKERS

| <u>Sample</u> | <u>Fiber (^w/_o)</u> | <u>Temperature (°F)</u> | <u>Time (Min.)</u> | <u>Average Tensile (psi)</u> |
|---------------|--|-----------------------------|------------------------|--------------------------------------|
| 881-34-1 | 8 | 2900 | 2 | 117,000 |
| 881-37-1 | 8 | 2900 | 4 | 87,400 |
| 881-38-2 | 8 | 2900 | 6 | 86,500 |
| 881-34-2 | 0 | 2900 | 2 | 83,000 |



Again, the fibers were left clean, and the metal formed small shot scattered throughout.

In neither case was there evidence of useful metal-fiber interaction.

II. Discussion

There is no doubt that useful metallurgical bonds between metal and alumina whiskers can be formed. This has been demonstrated with 80/20 nickel/chromium alloy, and repeatedly with iron. To generate such bonds it is necessary that oxygen be present at the metal-alumina interface, while the metal is in the molten state. It is further necessary that the amount of oxygen be limited to that which will react to form a strong bond; excessive amounts of oxygen lead to eventual destructive attack on the whiskers.

The addition of the needed oxygen through the surface of the metal melt has not succeeded because the molten metal and the fiber do not form a true interface, but separate at once, leaving the condition that fiber must be surrounded by molten metal unsatisfied.

Two methods for assuring the presence of the proper amount of oxygen are now under investigation. One method is to use the naturally-occurring oxides on the iron powder in the formation of the bond, and then remove the excess oxygen by the addition of aluminum metal. This requires a careful sequence of short-time heating followed by rapid addition of the aluminum.

The second method now under investigation is that of adding oxygen to the whiskers in the form of oxides of the matrix metal. The metal oxide is generated from a metal-organic compound applied to the whiskers in solution. After drying, the fibers are heat-treated in the presence of a stoichiometric



amount of oxygen to decompose the metal-organic and generate the desired oxide in situ.

In the case of iron, ferrous acetate appears to satisfy the requirements, being water-soluble, and decomposing readily to permit the generation of FeO or Fe₂O₃, depending upon the amount of oxygen supplied.

This method has the advantage that the bond-former is associated with the whisker itself, and may be controlled under less stringent conditions than those of the actual metal melt.

Concurrently, an investigation of possible bonding materials other than oxygen has been started. Additionally, equipment for fusing metal in vacuo by induction is now operative, and should yield information with respect to the effect of adsorbed contaminants on bond formation.

DISTRIBUTION LIST

1. Commander
Wright Air Development Division
Wright-Patterson Air Force Base, Ohio
Attn: ASRCNC
2. Brush Beryllium Company
4303 Perkins Avenue
Cleveland 3, Ohio
Attn: Mr. W. N. Beaver
3. Nuclear Metals, Incorporated
Concord, Massachusetts
Attn: Dr. A. Kaufman
4. Battelle Memorial Institute
505 King Avenue
Defense Metals Information Center
Columbus 1, Ohio
5. University of California
Lawrence Radiation Laboratory
P. O. Box 808
Livermore, California
Attn: Mr. Clovis C. Craig, Technical Information Div.
6. Lockheed Aircraft Corporation
Lockheed Missile System Division
Hanover Street
Palo Alto, California
7. Commander
Air Force Ballistic Missile Division
5760 Arbor Vitas Street
Inglewood 45, California
Attn: WDTLA
8. Avco Corporation
Research and Development Division
201 Lowell Street
Wilmington, Massachusetts
Attn: Dr. Thomas Vasilos
Chief, Metals and Ceramics

DISTRIBUTION LIST (cont'd.)

9. Commander
Watertown Arsenal
Watertown 72, Massachusetts
Attn: Mr. S. Arnold
10. Commander
Ordnance Corps. Frankford Arsenal
Pitman Dunn Laboratory
Philadelphia 37, Pennsylvania
Attn: Mr. D. Kleppinger
11. Chief of Naval Research
Department of the Navy
Washington 25, D. C.
Attn: ONR:423
12. Director, U. S. Naval Research Laboratory
Metallurgy Division
Washington 25, D. C.
Attn: Mr. W. Pellini
13. U. S. Atomic Energy Commission
Division of Reactor Development
Engineering Development Branch
Washington 25, D. C.
Attn: Mr. J. M. Simmons, Chief, Metallurgy Section
14. The Rand Corporation
Aeronautics Department
1700 Main Street
Santa Monica, California
Attn: Mr. George Hoffman
15. Department of the Navy
Bureau of Ships
Washington 25, D. C.
Attn: Code 343
16. Boeing Airplane Company
Seattle Division
Seattle, Washington
Attn: Mr. E. C. Bovee
Staff Eng'r. for Mtl's and Processes Staff

DISTRIBUTION LIST (cont'd.)

17. Republic Aviation Corporation
Farmingdale, Long Island, New York
Attn: Mr. Harry A. Pearl, Chief, Mtl's Development Div.
18. Metals Research Department
Armour Research Foundation
Chicago 16, Illinois
Attn: Dr. N. M. Parikh
19. Commander
Naval Air Material Center
Aeronautical Materials Laboratory
Philadelphia Naval Base
Philadelphia 12, Pennsylvania
20. National Aeronautics and Space Administration
1520 H Street, N. W.
Washington 25, D. C. (3 copies)
21. National Academy of Sciences
Materials Advisory Board
2101 Constitution Avenue
Washington 25, D. C.
Attn: Capt. A. M. Blamphin
22. National Beryllium Corporation
4501 Dell Avenue
North Bergen, New Jersey
Attn: Mr. C. E. Nelson
23. Clevite Corporation
Mechanical Research Division
540 East 105th Street
Cleveland 8, Ohio
Attn: Mr. A. D. Schwope
24. Narmco Industries Incorporated
Research and Development Division
8125 Aero Drive
San Diego 11, California
Attn: Mr. Finn Claudi-Magnussen
25. Sylvania-Corning Nuclear Corporation
Bayside, Long Island, New York
Attn: Mr. Charles I. Whitman

DISTRIBUTION LIST (cont'd.)

26. Southwest Research Institute
8500 Culebra Road
San Antonio 6, Texas
Attn: Dr. Tien-Shin Liu
27. Lockheed Aircraft Corporation
California Division
Burbank, California
Attn: Mr. E. A. Green
28. Diamond Ordnance Fuze Laboratories
Washington 25, D. C.
Attn: Mr. John Nusser, Engineers Tech.
Organization 930, Bldg. 16, Rm. 222
29. Department of the Interior
Bureau of Mines
Washington 25, D. C.
Attn: Mr. H. Austin Tucker
Ferrous Metals Branch
30. Dr. Eraldus Scala
Prof. Metallurgy and Materials Science
Cornell Aeronautical Laboratory
College of Engineering
Cornell University
Ithaca, New York
31. Mr. J. W. Weeton
Alloys and Composites Materials Branch
Materials Structures Division
Lewis Research Center
National Aeronautics and Space Adm.
Cleveland, Ohio
32. Department of the Navy
Bureau of Naval Weapons
Attn: RRMA-222
Washington 25, D. C.
33. Armed Services Technical Information Agency
Arlington Hall Station
Arlington, Virginia (8 copies)



DISTRIBUTION LIST (cont'd.)

34. Ordnance Corps
Watervliet Arsenal
Dr. Michael J. Salkind, SWEWV-RDR
Research Metallurgist
Watervliet, N. Y.
35. Mr. E. N. Petrick
Kelsey-Hayes Company
Romulus, Michigan
36. Inspector of Naval Materials
Ferguson Building
1783 East 11th Street
Cleveland 14, Ohio
Attn. Contract Control
37. Department of the Navy
Bureau of Naval Weapons
Attn. DLI-31
Washington 25, D. C.
38. Mr. J. I. Fisher
Huyck Metals
P O Box 30
Factory Lane
Milford, Connecticut

Mr. A. J. ...
Mr. N. A. ...
Mr. ... 41-90
Washington

