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14. ABSTRACT It has been shown that peridynamic simulation of damage process does not require any knowledge of the damage location and orientation prior to the simulation. This is fundamentally different from finite element analysis which requires knowledge of damage location and orientation in advance to impose special finite element mesh, such as initial damage elements and cohesive zone layers [1], for damage simulations. This prerequisite becomes even more challenging when inhomogeneous and anisotropic composite materials are of interest. In addition,					
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Report Title

Peridynamic Applications for Orthotropic Materials

ABSTRACT

It has been shown that peridynamic simulation of damage process does not require any knowledge of the damage location and orientation prior to the simulation. This is fundamentally different from finite element analysis which requires knowledge of damage location and orientation in advance to impose special finite element mesh, such as initial damage elements and cohesive zone layers [1], for damage simulations. This prerequisite becomes even more challenging when inhomogeneous and anisotropic composite materials are of interest. In addition, peridynamic simulation does not require remeshing at the end of each damage processing step since it is a mesh free method. On the contrary, finite element analysis does. Based on these difference, peridynamics should be more suitable for simulating dynamic damage process in composite materials which have different properties in different locations and different orientations.

Enter List of papers submitted or published that acknowledge ARO support from the start of the project to the date of this printing. List the papers, including journal references, in the following categories:

(a) Papers published in peer-reviewed journals (N/A for none)

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TOTAL:

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Liu, D., Jia, T. and Li, G., "Numerical Simulation of Q3D Composites for Blast Mitigation," Abstract, the 10th U.S. National Congress on Computational Mechanics, Columbus, OH, July 15-19, 2009.

Tao Jia and Dahsin Liu, "One-dimensional Peridynamic Analysis and Experimental Validation", the 11th US National Congress on Computational Mechanics, Minneapolis, MN, July 25-28, 2011.

Tao Jia, Development and applications of new peridynamic models, Ph.D Dissertation, Michigan State University, September, 2012.

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TOTAL:

Patents Submitted

Patents Awarded

Awards

none

Graduate Students

<u>NAME</u>	<u>PERCENT SUPPORTED</u>	Discipline
Tao Jia	0.50	
FTE Equivalent:	0.50	
Total Number:	1	

Names of Post Doctorates

<u>NAME</u>	<u>PERCENT SUPPORTED</u>
FTE Equivalent:	
Total Number:	

Names of Faculty Supported

<u>NAME</u>	<u>PERCENT SUPPORTED</u>	National Academy Member
Dahsin Liu	0.20	
FTE Equivalent:	0.20	
Total Number:	1	

Names of Under Graduate students supported

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Names of Personnel receiving masters degrees

<u>NAME</u>
Total Number:

Names of personnel receiving PHDs

<u>NAME</u>
Tao Jia
Total Number:

Names of other research staff

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Sub Contractors (DD882)

Inventions (DD882)

Scientific Progress

An orthotropic material model was presented and its applications were studied in this report. A four-parameter peridynamic model for orthotropic materials was proposed to coincide with the four independent material properties required for orthotropic materials. This four-parameter model was different from other models as it had four parameters and was mesh independent. The model was verified by a static tensile test and a vibration excitation of a laminated beam. An SEN (single edge notch) test of a 0° laminated plate was simulated by peridynamics and the computational results matched very well with published experimental results. Fracture initiation and crack path of laminated plates with different fiber orientations were also studied using peridynamics. The mesh-free peridynamic model was convenient and efficient since there was no need to have different meshes for different fiber orientations. Without using special meshes and requiring prior knowledge of fracture paths, a general peridynamic code was able to predict fracture velocity and crack path successfully.

Technology Transfer

Peridynamic Applications for Orthotropic Materials

It has been shown earlier that peridynamic simulation of damage process does not require any knowledge of the damage location and orientation prior to the simulation. This is fundamentally different from finite element analysis which requires knowledge of damage location and orientation in advance to impose special finite element mesh, such as initial damage elements and cohesive zone layers [1], for damage simulations. This prerequisite becomes even more challenging when inhomogeneous and anisotropic composite materials are of interest. In addition, peridynamic simulation does not require remeshing at the end of each damage processing step since it is a mesh free method. On the contrary, finite element analysis does. Based on these difference, peridynamics should be more suitable for simulating dynamic damage process in composite materials which have different properties in different locations and different orientations.

Quite some simulations of composite damage process have been available in the literatures. Dwivedi [1] modeled the propagation of single-edge notch (SEN) in 0° laminated plate using cohesive zone method. Xu [2] and Hu [3] proposed a two-parameter discrete peridynamic model for composite damage simulations, in which there were two kinds of bonds: fiber bond and matrix bond. Two material properties, a_1 and a_2 , were associated with the two types of bonds. Only the bonds along the fiber direction were associated with the material property a_1 while all other bonds with the material property a_2 . This model required remeshing for different fiber directions. For example, a 0° - 90° grid mesh could only be used for a 0° or 90° laminae. For

a 45° lamina, a grid mesh consisting of 45° and 135° was required. The two-parameter model was an approximation of the four material properties involved in orthotropic materials. They were mainly associated with two Young's moduli, E_1 and E_2 . Its capability of modeling shear behavior is unknown.

In this study, a continuous orthotropic material model is proposed. It is based on continuous trigonometric functions. With the continuous material property functions, it is not necessary to have bond in fiber direction and therefore, this model is mesh independent.

5.1 Bar model for orthotropic materials

This model is based on the bar model presented in Section 4.1. The peridynamic equation of motion [4] in two-dimensional domain can be expressed as

$$\rho \ddot{\mathbf{u}} = \int \mathbf{f} dA + \mathbf{b} \quad (5.1)$$

where \mathbf{b} is external force. The force boundary condition can be included in the external force.

For bar model, the bond function f is

$$f = c \cdot s \quad (5.2)$$

where s is bond stretch. Contrary to the isotropic material model, bond material property c is assumed to be a trigonometric function

$$c = d_1 \cos(\theta - \alpha)^4 + d_2 \cos(\theta - \alpha)^2 + d_3 \quad (5.3)$$

where d_1 , d_2 and d_3 are constants and can be identified from composite material properties. θ is bond direction and α is the fiber direction as shown in Fig. 5.1.

Similar to the analysis in the previous chapters, d_1 , d_2 and d_3 can be identified from comparing the strain energy densities based on peridynamic analysis and those based on classical mechanics.

Consider a composite plate with the fibers oriented in α° direction and subjected to the following strain field

$$\varepsilon_{xx} = \varepsilon_1 \quad (5.4)$$

$$\varepsilon_{yy} = \varepsilon_2 \quad (5.5)$$

$$\gamma_{xy} = \gamma_{12} \quad (5.6)$$

The three components are independent of one another.

For a bond in θ direction, and connected to a point \mathbf{x} in the domain, the bond force should be

$$f = c \cdot (\varepsilon_1 \cos^2 \theta + \varepsilon_2 \sin^2 \theta + \gamma_{12} \sin \theta \cos \theta) \quad (5.7)$$

From Eqn. 4.7, the strain energy in the bond becomes

$$w_b = \frac{c\eta^2}{2\xi} = cs^2\xi/2 \quad (5.8)$$

Substituting Eqn. 5.3 and Eqn. 5.7 into Eqn. 5.8 and integrating w_b over the horizon, the strain energy density at the point \mathbf{x} should be

$$W = \frac{1}{2} \int w_b dA = \frac{1}{2} \int_0^\delta \int_0^{2\pi} \frac{cs^2\xi}{2} \xi d\theta d\xi = \delta^3 \pi \{16(d_1 + d_2)(\varepsilon_1 - \varepsilon_2)(\varepsilon_1 + \varepsilon_2) \cos 2\alpha + d_1[(\varepsilon_1 - \varepsilon_2)^2 - \gamma_{12}^2] \cos 4\alpha + 2[(3d_1 + 4d_2 + 8d_3)(3\varepsilon_1^2 + 2\varepsilon_1\varepsilon_2 + 3\varepsilon_2^2 + \gamma_{12}^2) + 8(d_1 + d_2)(\varepsilon_1 + \varepsilon_2)\gamma_{12} \sin 2\alpha + d_1(\varepsilon_1 - \varepsilon_2)\gamma_{12} \sin 4\alpha]\}/768 \quad (5.9)$$

Eqn. 5.9 can be simplified to find the coefficient of each independent term, as shown in Table 5.1. The simplification is achieved based on Mathematica [5].

Table 5.1 Simplified Eqn. 5.9 in terms of independent terms

Coefficients	Independent terms
$d_1\delta^3\pi/96$	$\cos \alpha^4 \varepsilon_1^2$
$d_1\delta^3\pi/96$	$\cos \alpha^4 \varepsilon_2^2$
$-d_1\delta^3\pi/48$	$\cos \alpha^4 \varepsilon_1\varepsilon_2$
$-d_1\delta^3\pi/96$	$\cos \alpha^4 \gamma_{12}^2$
$(3d_1 + 4d_2)\delta^3\pi/96$	$\cos \alpha^2 \varepsilon_1^2$
$(-5d_1 - 4d_2)\delta^3\pi/96$	$\cos \alpha^2 \varepsilon_2^2$
$d_1\delta^3\pi/48$	$\cos \alpha^2 \varepsilon_1\varepsilon_2$
$d_1\delta^3\pi/96$	$\cos \alpha^2 \gamma_{12}^2$
$(3d_1 + 8d_2 + 48d_3)\delta^3\pi/768$	ε_1^2
$(35d_1 + 40d_2 + 48d_3)\delta^3\pi/768$	ε_2^2
$(5d_1 + 8d_2 + 16d_3)\delta^3\pi/384$	$\varepsilon_1\varepsilon_2$
$(5d_1 + 8d_2 + 16d_3)\delta^3\pi/768$	γ_{12}^2
$(3d_1 + 4d_2)\delta^3\pi/96$	$\sin \alpha \cos \alpha \gamma_{12}\varepsilon_1$
$(5d_1 + 4d_2)\delta^3\pi/96$	$\sin \alpha \cos \alpha \gamma_{12}\varepsilon_2$
$d_1\delta^3\pi/48$	$\sin \alpha \cos \alpha^3 \gamma_{12}\varepsilon_1$
$-d_1\delta^3\pi/48$	$\sin \alpha \cos \alpha^3 \gamma_{12}\varepsilon_2$

On the other hand, from the theory of composite materials [6], the strain energy density under

$\varepsilon_{xx} = \varepsilon_1$, $\varepsilon_{yy} = \varepsilon_2$ and $\gamma_{xy} = \gamma_{12}$ is

$$W = \frac{1}{2}\sigma_{xx}\varepsilon_1 + \frac{1}{2}\sigma_{yy}\varepsilon_2 + \frac{1}{2}\sigma_{xy}\gamma_{12} \quad (5.10)$$

The stresses in Eqn. 5.10 can be calculated by

$$\begin{bmatrix} \sigma_{xx} \\ \sigma_{yy} \\ \sigma_{xy} \end{bmatrix} = \begin{bmatrix} Q_{xx} & Q_{xy} & Q_{xs} \\ Q_{xy} & Q_{yy} & Q_{ys} \\ Q_{xs} & Q_{ys} & Q_{ss} \end{bmatrix} \begin{bmatrix} \varepsilon_{xx} \\ \varepsilon_{yy} \\ \gamma_{xy} \end{bmatrix} \quad (5.11)$$

where,

$$Q_{xx} = Q_{11}m^4 + Q_{22}n^4 + 2m^2n^2Q_{12} + 4m^2n^2Q_{66} \quad (5.12)$$

$$Q_{yy} = Q_{11}n^4 + Q_{22}m^4 + 2m^2n^2Q_{12} + 4m^2n^2Q_{66} \quad (5.13)$$

$$Q_{xy} = Q_{11}m^2n^2 + Q_{22}m^2n^2 + (m^4 + n^4)Q_{12} - 4m^2n^2Q_{66} \quad (5.14)$$

$$Q_{xs} = m^3nQ_{11} - mn^3Q_{22} - mn(m^2 - n^2)Q_{12} - 2mn(m^2 - n^2)Q_{66} \quad (5.15)$$

$$Q_{ys} = mn^3Q_{11} - m^3nQ_{22} + mn(m^2 - n^2)Q_{12} + 2mn(m^2 - n^2)Q_{66} \quad (5.16)$$

$$Q_{ss} = m^2n^2Q_{11} + m^2n^2Q_{22} - 2m^2n^2Q_{12} + (m^2 - n^2)^2Q_{66} \quad (5.17)$$

$$m = \cos \alpha \quad (5.18)$$

$$n = \sin \alpha \quad (5.19)$$

$$Q_{11} = \frac{E_1}{1 - \nu_{12}\nu_{21}} \quad (5.20)$$

$$Q_{22} = \frac{E_2}{1 - \nu_{12}\nu_{21}} \quad (5.21)$$

$$Q_{12} = \frac{E_2\nu_{12}}{1 - \nu_{12}\nu_{21}} \quad (5.22)$$

$$Q_{66} = G_{12} \quad (5.23)$$

E_1, E_2, ν_{12} and G_{12} are the four material properties of orthotropic materials. Substituting Eqns. 5.11 - 5.23 into Eqn. 5.10, it yields

$$\begin{aligned} W = & \frac{1}{2(E_1 - E_2\nu_{12}^2)} \{ (\varepsilon_1^2 E_1^2 + E_1 \varepsilon_2^2 E_2 + E_1 G_{12} \gamma_{12}^2 + 2\varepsilon_1 E_1 \varepsilon_2 E_2 \nu_{12} - E_2 G_{12} \gamma_{12}^2 \nu_{12}^2) \cos \alpha^4 + \\ & 2\gamma_{12} [\varepsilon_2 (E_1 (2G_{12} + E_2 (-1 + \nu_{12})) - 2E_2 G_{12} \nu_{12}^2) + \varepsilon_1 (E_1^2 + 2E_2 G_{12} \nu_{12}^2 - E_1 (2G_{12} + \\ & E_2 \nu_{12}))] \cos \alpha^3 \sin \alpha + [2\varepsilon_1 E_1 \varepsilon_2 (E_1 + E_2) + E_1^2 \gamma_{12}^2 + 2\varepsilon_1^2 E_1 E_2 \nu_{12} + 2E_2 G_{12} \gamma_{12}^2 \nu_{12}^2 + \\ & E_1 (-2G_{12} \gamma_{12}^2 + E_2 (\gamma_{12}^2 + 2\varepsilon_2^2 \nu_{12} - 2\gamma_{12}^2 \nu_{12}))] \cos \alpha^2 \sin \alpha^2 + 2\gamma_{12} [E_1^2 \varepsilon_2 + 2(-\varepsilon_1 + \\ & \varepsilon_2) E_2 G_{12} \nu_{12}^2 + E_1 (\varepsilon_1 (2G_{12} + E_2 (-1 + \nu_{12})) - \varepsilon_2 (2G_{12} + E_2 \nu_{12}))] \cos \alpha \sin \alpha^3 + [E_1^2 \varepsilon_2^2 - \\ & E_2 G_{12} \gamma_{12}^2 \nu_{12}^2 + E_1 (\varepsilon_1^2 E_2 + G_{12} \gamma_{12}^2 + 2\varepsilon_1 \varepsilon_2 E_2 \nu_{12})] \sin \alpha^4 + (\varepsilon_1 - \varepsilon_2)^2 G_{12} (E_1 - E_2 \nu_{12}^2) \sin 2\alpha^2 \} \end{aligned} \quad (5.24)$$

After simplification processes, the coefficient of each independent terms can be found. They are listed in Table 5.2.

Table 5.2 Simplified Eqn. 5.12 in terms of independent terms

Coefficients	Independent terms
$\frac{E_1^2 + 4E_2 G_{12} \nu_{12}^2 + E_1 (E_2 - 4G_{12} - 2E_2 \nu_{12})}{2(E_1 - E_2 \nu_{12}^2)}$	$\cos \alpha^4 \varepsilon_1^2$
$\frac{E_1^2 + 4E_2 G_{12} \nu_{12}^2 + E_1 (E_2 - 4G_{12} - 2E_2 \nu_{12})}{2(E_1 - E_2 \nu_{12}^2)}$	$\cos \alpha^4 \varepsilon_2^2$
$-\frac{E_1^2 + 4E_2 G_{12} \nu_{12}^2 + E_1 (E_2 - 4G_{12} - 2E_2 \nu_{12})}{(E_1 - E_2 \nu_{12}^2)}$	$\cos \alpha^4 \varepsilon_1 \varepsilon_2$

$-\frac{E_1^2 + 4E_2G_{12}v_{12}^2 + E_1(E_2 - 4G_{12} - 2E_2v_{12})}{2(E_1 - E_2v_{12}^2)}$	$\cos \alpha^4 \gamma_{12}^2$
$\frac{E_1(2G_{12} + E_2(-1 + v_{12})) - 2E_2G_{12}v_{12}^2}{E_1 - E_2v_{12}^2}$	$\cos \alpha^2 \varepsilon_1^2$
$\frac{-E_1^2 + 2E_1G_{12} + E_1E_2v_{12} - 2E_2G_{12}v_{12}^2}{E_1 - E_2v_{12}^2}$	$\cos \alpha^2 \varepsilon_2^2$
$\frac{E_1^2 + 4E_2G_{12}v_{12}^2 + E_1(E_2 - 4G_{12} - 2E_2v_{12})}{E_1 - E_2v_{12}^2}$	$\cos \alpha^2 \varepsilon_1\varepsilon_2$
$\frac{E_1^2 + 4E_2G_{12}v_{12}^2 + E_1(E_2 - 4G_{12} - 2E_2v_{12})}{2(E_1 - E_2v_{12}^2)}$	$\cos \alpha^2 \gamma_{12}^2$
$\frac{E_1E_2}{2(E_1 - E_2v_{12}^2)}$	ε_1^2
$\frac{E_1^2}{2(E_1 - E_2v_{12}^2)}$	ε_2^2
$\frac{E_1E_2v_{12}}{E_1 - E_2v_{12}^2}$	$\varepsilon_1\varepsilon_2$
$\frac{E_1G_{12} - E_2G_{12}v_{12}^2}{2(E_1 - E_2v_{12}^2)}$	γ_{12}^2
$\frac{E_1(2G_{12} + E_2(-1 + v_{12})) - 2E_2G_{12}v_{12}^2}{E_1 - E_2v_{12}^2}$	$\sin \alpha \cos \alpha \gamma_{12}\varepsilon_1$
$\frac{E_1^2 + 2E_2G_{12}v_{12}^2 + E_1(-2G_{12} - E_2v_{12})}{E_1 - E_2v_{12}^2}$	$\sin \alpha \cos \alpha \gamma_{12}\varepsilon_2$
$\frac{E_1^2 + 4E_2G_{12}v_{12}^2 + E_1(E_2 - 4G_{12} - 2E_2v_{12})}{(E_1 - E_2v_{12}^2)}$	$\sin \alpha \cos \alpha^3 \gamma_{12}\varepsilon_1$
$-\frac{E_1^2 + 4E_2G_{12}v_{12}^2 + E_1(E_2 - 4G_{12} - 2E_2v_{12})}{(E_1 - E_2v_{12}^2)}$	$\sin \alpha \cos \alpha^3 \gamma_{12}\varepsilon_2$

Eqn. 5.9 should be equal to Eqn. 5.24. Hence, it is possible to set the coefficients of identical

terms equal to each other and identify d_1 , d_2 and d_3 .

$$d_1 = \frac{24}{\delta^3 E_1 \pi} \left(E_1^2 + E_1 E_2 - 12 E_1 G_{12} + \frac{E_1 \sqrt{E_1 E_2 + 4 G_{12}^2}}{\sqrt{\frac{E_2}{E_1}}} + \sqrt{E_1 E_2} \sqrt{E_1 E_2 + 4 G_{12}^2} \right) \quad (5.25)$$

$$d_2 = \frac{6}{\delta^3 E_1 \pi} \left(-3 E_1^2 - 5 E_1 E_2 + 48 E_1 G_{12} - \frac{3 E_1 \sqrt{E_1 E_2 + 4 G_{12}^2}}{\sqrt{\frac{E_2}{E_1}}} - 5 \sqrt{E_1 E_2} \sqrt{E_1 E_2 + 4 G_{12}^2} \right) \quad (5.26)$$

$$d_3 = \frac{3}{2 \delta^3 E_1 \pi} \left(E_1^2 + 5 E_1 E_2 - 20 E_1 G_{12} + \frac{E_1 \sqrt{E_1 E_2 + 4 G_{12}^2}}{\sqrt{E_2/E_1}} + 5 \sqrt{E_1 E_2} \sqrt{E_1 E_2 + 4 G_{12}^2} \right) \quad (5.27)$$

$$v_{12} = \frac{-E_1 E_2 + \sqrt{E_1 E_2} \sqrt{E_1 E_2 + 4 G_{12}^2}}{2 E_2 G_{12}} \quad (5.28)$$

The composite model has three parameters and it can model shearing deformation. The bond stiffness is a continuous function so the stiffness at any direction can be calculated. It can be found that the bond becomes stronger as the angle between the bond and the fiber becomes smaller. Similar to the bar model in Section 4.1, the Poisson's ratio of this model is fixed and is related to the remaining three independent material properties. For example, the Poisson's ratio of Kevlar/Epoxy is 0.34 while the Poisson's ratio calculated from this model is 0.39.

5.2 Beam model for orthotropic materials

To accommodate the four material properties of orthotropic materials, a beam model based on Section 4.2 is proposed here.

From the composite theory, there are four independent material properties, E_1 , E_2 , v_{12} and G_{12} .

The composite stiffness varies with fiber orientation. Bond functions are

$$f'_x = \frac{c_1(u'_2 - u'_1)}{r} \quad (5.29)$$

$$f'_y = \frac{c_2(v'_2 - v'_1)}{r^3} \quad (5.30)$$

They are identical to Eqn. 4.23 and Eqn. 4.24. However, c_1 and c_2 for a bond in θ direction will be dependent on fiber orientations as follows,

$$c_1 = d_1 \cos(\theta - \alpha)^4 + d_2 \cos(\theta - \alpha)^2 + d_3 \quad (5.31)$$

$$c_2 = d_4 \quad (5.32)$$

where α is the orientation of the fiber, as shown in Fig. 5.3. The coefficients d_1 , d_2 , d_3 and d_4 are four independent material properties. They are related to the four material properties defined in the composite theory.

Consider a composite plate with fiber in α direction and subject to the following strain field

$$\varepsilon_{xx} = \varepsilon_1 \quad (5.33)$$

$$\varepsilon_{yy} = \varepsilon_2 \quad (5.34)$$

$$\gamma_{xy} = \gamma_{12} \quad (5.35)$$

The three strains are independent from each other.

Similar to Eqn. 4.29, the strain energy in one bond becomes

$$w_b = \frac{c_1(\varepsilon_1 r \cos \theta^2 + \varepsilon_2 r \sin \theta^2)^2}{2r} + \frac{c_2(-\varepsilon_1 r \cos \theta \sin \theta + \varepsilon_2 r \cos \theta \sin \theta)^2}{2r^3} \quad (5.36)$$

Integrate Eqn. 5.36 to find the strain energy density for a point

$$W = \frac{1}{2} \int w_b dA = \frac{1}{2} \int_0^\delta \int_0^{2\pi} w_b r d\theta dr = \frac{1}{768} \pi \delta \{16(d_1 + d_2) \delta^2 (\varepsilon_1^2 - \varepsilon_2^2) \cos 2\alpha +$$

$$\begin{aligned}
& d_1 \delta^2 (\varepsilon_1^2 - 2\varepsilon_1 \varepsilon_2 + \varepsilon_2^2 - \gamma_{12}^2) \cos 4\alpha + 2[24d_4 (\varepsilon_1 - \varepsilon_2)^2 + 9d_1 \delta^2 \varepsilon_1^2 + 12d_2 \delta^2 \varepsilon_1^2 + \\
& 24d_3 \delta^2 \varepsilon_1^2 + 6d_1 \delta^2 \varepsilon_1 \varepsilon_2 + 8d_2 \delta^2 \varepsilon_1 \varepsilon_2 + 16d_3 \delta^2 \varepsilon_1 \varepsilon_2 + 9d_1 \delta^2 \varepsilon_2^2 + 12d_2 \delta^2 \varepsilon_2^2 + 24d_3 \delta^2 \varepsilon_2^2 + \\
& 24d_4 \gamma_{12}^2 + 3d_1 \delta^2 \gamma_{12}^2 + 4d_2 \delta^2 \gamma_{12}^2 + 8d_3 \delta^2 \gamma_{12}^2 + 8(d_1 + d_2) \delta^2 (\varepsilon_1 + \varepsilon_2) \gamma_{12} \sin 2\alpha + \\
& d_1 \delta^2 (\varepsilon_1 - \varepsilon_2) \gamma_{12} \sin 4\alpha] \} \tag{5.37}
\end{aligned}$$

The Strain energy density based on the composite theory is the same as Eqn. 5.24. Similar to Section 5.1, by setting Eqn. 5.24 equal to Eqn. 5.37 and comparing the coefficients of the independent terms, the following equations are obtained

$$d_1 = \frac{48(E_1^2 + E_1 E_2 - 4E_1 G_{12} - 2E_1 E_2 \nu_{12} + 4E_2 G_{12} \nu_{12}^2)}{\pi(E_1 - E_2 \nu_{12}^2) \delta^3} \tag{5.38}$$

$$d_2 = -\frac{12(3E_1^2 + 5E_1 E_2 - 16E_1 G_{12} - 8E_1 E_2 \nu_{12} + 16E_2 G_{12} \nu_{12}^2)}{\pi(E_1 - E_2 \nu_{12}^2) \delta^3} \tag{5.39}$$

$$d_3 = \frac{3(E_1^2 + 5E_1 E_2 - 8E_1 G_{12} - 2E_1 E_2 \nu_{12} + 8E_2 G_{12} \nu_{12}^2)}{\pi(E_1 - E_2 \nu_{12}^2) \delta^3} \tag{5.40}$$

$$d_4 = -\frac{4(-E_1 G_{12} + E_1 E_2 \nu_{12} + E_2 G_{12} \nu_{12}^2)}{\pi(E_1 - E_2 \nu_{12}^2) \delta} \tag{5.41}$$

5.3 Calculation of stresses from peridynamics

Similar to Section 4.2, stresses can be defined and calculated from peridynamics. They can be used for some special comparison but not necessary in peridynamic simulations.

Consider a case with $\varepsilon_{xx} = \varepsilon_1$ and $\varepsilon_{yy} = \varepsilon_2$. The stress σ_{xx} can be calculated by using Eqn. 4.39 and is given below

$$\sigma_{xx} = \int_0^\delta \int_t^\delta \int_{-\cos^{-1}\frac{t}{r}}^{\cos^{-1}\frac{t}{r}} (fx' \cdot \cos \theta - fy' \cdot \sin \theta) r \, d\theta dr dt = \frac{1}{384} \pi \delta \{48d_4 (\varepsilon_1 - \varepsilon_2) +$$

$$2(3d_1 + 4d_2 + 8d_3)(3\varepsilon_1 + \varepsilon_2)\delta^2 + \delta^2[16\varepsilon_1 \cos 2\alpha (d_1 + d_2) + d_1(\varepsilon_1 - \varepsilon_2) \cos 4\alpha] \quad (5.42)$$

Substituting Eqns. 5.38 - 5.41 into Eqn. 5.42, it yields

$$\begin{aligned} \sigma_{xx} = & \\ & \frac{1}{8(E_1 - E_2 v_{12}^2)} \{ 3\varepsilon_1 E_1^2 + \varepsilon_2 E_1^2 + 3\varepsilon_1 E_1 E_2 + \varepsilon_2 E_1 E_2 + 4\varepsilon_1 E_1 G_{12} - 4\varepsilon_2 E_1 G_{12} + 2\varepsilon_1 E_1 E_2 v_{12} + \\ & 6\varepsilon_2 E_1 E_2 v_{12} - 4\varepsilon_1 E_2 G_{12} v_{12}^2 + 4\varepsilon_2 E_2 G_{12} v_{12}^2 + 4\varepsilon_1 E_1 (E_1 - E_2) \cos 2\alpha + (\varepsilon_1 - \varepsilon_2)(E_1^2 + \\ & 4E_2 G_{12} v_{12}^2 + E_1(E_2 - 4G_{12} - 2E_2 v_{12})) \cos 4\alpha \} \end{aligned} \quad (5.43)$$

While from the composite theory, σ_{xx} can be expressed as

$$\begin{aligned} \sigma_{xx} = Q_{xx}\varepsilon_1 + Q_{xy}\varepsilon_2 = & \frac{1}{(E_1 - E_2 v_{12}^2)} \{ E_1(\varepsilon_1 E_1 + \varepsilon_2 E_2 v_{12}) \cos^4 \alpha + [E_1^2 \varepsilon_2 + 4(-\varepsilon_1 + \\ & \varepsilon_2)E_2 G_{12} v_{12}^2 + E_1(\varepsilon_2 E_2 + 4\varepsilon_1 G_{12} - 4\varepsilon_2 G_{12} + 2\varepsilon_1 E_2 v_{12})] \cos^2 \alpha \sin^2 \alpha + E_1 E_2 (\varepsilon_1 + \\ & \varepsilon_2 v_{12}) \sin^4 \alpha \} \end{aligned} \quad (5.44)$$

With further simplification, it can be found that Eqn. 5.43 is identical to Eqn. 5.44.

5.4 Laminated plate under static loading

In this section, it is to verify the proposed peridynamic model with an analytical solution. A simple tensile test is performed on a laminated plate. The peridynamic results will be compared with the results obtained from the composite theory.

Consider a $100 \text{ mm} \times 100 \text{ mm}$ laminated plate with fibers in α direction as shown in Fig. 5.4. A tensile pressure of 10 MPa is applied at the bottom and the top of the plate. The plate is made of

E-Glass/Epoxy and the material properties are shown in Table 5.3.

Table 5.3 Material properties of E-Glass/Epoxy

Longitudinal Young's modulus, E_1	41 GPa
Transverse Young's modulus, E_2	10.4 GPa
Poisson's ratio, ν_{12}	0.28
Shear modulus, G_{12}	4.3 GPa
Mass density, ρ	1970 kg/m ³

Based on the composite theory [6], the components of the compliance matrix are

$$S_{11} = 1/E_1 \quad (5.45)$$

$$S_{22} = 1/E_2 \quad (5.46)$$

$$S_{12} = -\nu_{12}/E_1 \quad (5.47)$$

$$S_{66} = 1/G_{12} \quad (5.48)$$

Strains from the composite theory can be calculated as follows

$$\begin{bmatrix} \varepsilon_{xx} \\ \varepsilon_{yy} \\ \gamma_{xy} \end{bmatrix} = \begin{bmatrix} S_{xx} & S_{xy} & S_{xs} \\ S_{xy} & S_{yy} & S_{ys} \\ S_{xs} & S_{ys} & S_{ss} \end{bmatrix} \begin{bmatrix} \sigma_{xx} \\ \sigma_{yy} \\ \sigma_{xy} \end{bmatrix} \quad (5.49)$$

The transformed compliance matrix is calculated as

$$\begin{bmatrix} S_{xx} & S_{xy} & S_{xs} \\ S_{xy} & S_{yy} & S_{ys} \\ \frac{1}{2}S_{xs} & \frac{1}{2}S_{ys} & \frac{1}{2}S_{ss} \end{bmatrix} = T^{-1} \begin{bmatrix} S_{11} & S_{12} & 0 \\ S_{12} & S_{22} & 0 \\ 0 & 0 & \frac{1}{2}S_{66} \end{bmatrix} T \quad (5.50)$$

where

$$T = \begin{bmatrix} m^2 & n^2 & 2mn \\ n^2 & m^2 & -2mn \\ -mn & mn & m^2 - n^2 \end{bmatrix} \quad (5.51)$$

and $m = \cos \alpha$ and $n = \sin \alpha$.

From Eqn. 5.49, the displacement field can be obtained from the composite theory. Displacements from peridynamics are compared with those from the composite theory in Fig. 5.5-Fig. 5.10 with $\alpha = 0^\circ, 45^\circ$ and 60° . As can be seen, the results from peridynamics and those from composite theory are identical to each other.

5.5 Free vibration of a laminated beam

The free vibration of a laminated beam is investigated in this section by the Classical Laminated Beam Theory and the solution will be used to verify that obtained from peridynamic model.

Consider a simply supported beam as shown Fig. 5.11. The length of the beam is $L = a \cdot h$ and the thickness of the beam is $h = 10 \text{ mm}$, where a is the aspect ratio of the beam. The beam is made of Kevlar/Epoxy with fibers oriented in x direction. The material properties are shown in Table 5.4.

Table 5.4 Material properties of Kevlar/Epoxy

Longitudinal Young's modulus, E_1	80 GPa
Transverse Young's modulus, E_2	5.5 GPa
Poisson's ratio, ν_{12}	0.34
Shear modulus, G_{12}	2.2 GPa

Mass density, ρ	1380 kg/m ³
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From [7], the governing equations of the beam are

$$A_{11} \frac{d^2 u^o}{dx^2} - B_{11} \frac{d^3 w}{dx^3} - \rho \omega^2 u^o = 0 \quad (5.52)$$

$$D_{11} \frac{d^4 w}{dx^4} - B_{11} \frac{d^3 u^o}{dx^3} - \rho \omega^2 w = 0 \quad (5.53)$$

where

$$A_{11} = \int_{-h/2}^{h/2} Q_{11} dz = Q_{11} h \quad (5.54)$$

$$B_{11} = \int_{-h/2}^{h/2} Q_{11} z dz = 0 \quad (5.55)$$

$$D_{11} = \int_{-h/2}^{h/2} Q_{11} z^2 dz = Q_{11} h^3 / 12 \quad (5.56)$$

A solution satisfying the governing equations is

$$u_0 = U \cos px \quad (5.57)$$

$$w = W \sin px \quad (5.58)$$

Eqn. 5.57 and Eqn. 5.58 satisfy the simply supported boundary conditions automatically.

Substituting Eqn. 5.57 and Eqn. 5.58 into Eqn. 5.52 and Eqn. 5.53, it yields

$$-A_{11} p^2 U - \rho \omega^2 U = 0 \quad (5.59)$$

$$D_{11} p^4 W - \rho \omega^2 W = 0 \quad (5.60)$$

Solving Eqn. 5.59 and Eqn. 5.60, it can be concluded that

$$\omega^2 = \frac{Q_{11} p^4 h^3}{12\rho} \quad (5.61)$$

$$U = 0 \quad (5.62)$$

If the initial condition of the beam is

$$w(t = 0) = 1 \times 10^{-5} \sin \frac{\pi}{L} x \quad (5.63)$$

then

$$W = 1 \times 10^{-5} \quad (5.64)$$

$$p = \frac{\pi}{L} \quad (5.65)$$

The solution of the problem should have the following forms

$$w'(x, t) = W \sin px \cos \omega t \quad (5.66)$$

$$u'(x, z, t) = -zpW \cos px \cos \omega t \quad (5.67)$$

where W , p and ω can be found from Eqn. 5.64, Eqn. 5.65 and Eqn. 5.61, respectively.

The same problem can be simulated by peridynamics. The displacement history of point A and point B (Fig. 5.11) are recorded. Fig. 5.12 compares the peridynamic results with those from the composite beam theory for the aspect ratio $a = 5$. Fig. 5.13 compares the peridynamic results with those from the composite beam theory for the aspect ratio $a = 20$. Results from the two methods show good match in vertical displacement w of point B. For horizontal displacement u of point A, peridynamic result is almost the same as the beam theory result when the aspect ratio $a = 20$. However, there is difference between the peridynamic result and the beam theory result when the aspect ratio $a = 5$. This is because the beam theory assumes no variation of vertical displacement when the beam is slender. With a small aspect ratio, such as $a = 5$, this variation is not negligible and the beam theory does not provide a good approximation.

5.6 Comparisons of crack propagation velocity with experiments

Experimental studies on dynamic crack propagation in fiber-reinforced composite materials have been conducted by Zheng[9], Rosakis[8], Stout[10] and Coker[11,12]. It has been shown that a weak fracture plane usually occurs between fiber and matrix in unidirectional fiber-reinforced composites. Due to material anisotropy, the wave speed along the fiber direction is very different from that along the perpendicular direction. Dynamic crack propagation has been commonly investigated by finite element method. A limited number of computational studies have been reported by Huang [13], Hwang [14], Kumar [15], Stout [10], Lo [16], Sun [17] and Pandey[18]. The limit of computational works is likely due to the requirement of remeshing and the complexity of handling elements once crack starts. In this section, dynamic crack propagation in a unidirectional graphite/epoxy composite is studied with the use of peridynamics. The computational results are validated by the experimental results given in reference [8].

Consider a $76\text{ mm} \times 152\text{ mm}$ unidirectional graphite/epoxy fiber-reinforced composite plates under three-point bending [8] as shown in Fig. 5.14. The fiber is in 0° direction. The material properties are shown in Table 5.5 [8, 22]. There is a notch with a length of 15.2 mm , i.e. 20% of the plate width, at the left boundary of the plate. This crack length is used because it was used in the past to produce reliable results in dynamic fracture experiments [19]. To minimize residual stresses due to machining, a low-speed diamond saw was used to produce the initial notch with a width of approximately 1.5 mm .

Table 5.5 Material properties of graphite/epoxy

Longitudinal Young's modulus, E_1	150 GPa
Transverse Young's modulus, E_2	11.6 GPa
Poisson's ratio, ν_{12}	0.36
Shear modulus, G_{12}	3.5 GPa
Mode I intralaminar fracture energy for longitudinal loading, G_{10}	77.9 J/m ²
Mode I intralaminar fracture energy for transver loading, G_{20}	5 J/m ²
Mass density, ρ	1590 kg/m ³

A drop-weight tower is used to introduce impact on the opposite side of the notch with an impacting speed of $v_0 = 4 \text{ m/s}$. After the impact, stress waves propagate to the interior of the plate and then reflects from the boundaries. Because of the anisotropy of the material, stress waves travel in different directions at different velocities. Experiments show that the crack starts to propagate at about $25 \mu\text{s}$ after the impact so the effects of dispersion are not very important since the applied stress pulse is very long (about $120 \mu\text{s}$) compared with the time of crack initiation. Therefore, loading is continuously applied throughout the entire event.

The real-time visualization of dynamic fracture is produced by an optical method of coherent gradient sensing (CGS) in reflection [20, 21]. Imaging is performed with a rotating-mirror high-speed camera. Details of the CGS system can be found in [8, 20, 21].

Fig. 5.15 shows the crack propagation velocity from the peridynamic simulation. The initial time $t = 0$ is used to denote the beginning of the crack propagation. For negative time, $v = 0 \text{ m/s}$. The crack starts from about 700 m/s and accelerates to 900 m/s within about $10 \mu\text{s}$. It then

decelerates to less than 500 m/s in $40\ \mu\text{s}$ after the initiation. This deceleration is believed to be due to the fact that the growing crack tip enters a region of high compressive stress zone as it approaches the loading area. The peridynamic computational results are compared with the experimental results from Ref. [8]. As shown in Fig. 5.15, they agree each other reasonably well.

5.7 Dynamic fracture mode in unidirectional composites

In order to investigate the behavior of cracks, Wu [24] conducted experiments with unidirectional, fiberglass-reinforced Scotch composites with a centered precrack in the direction of fibers. The composites were loaded with tension, pure shear and combined tension and shear. In all three cases, it was observed that the crack propagated in the same direction as the fiber direction. Finite element analysis were also used to study the damage path and failure initiation of pre-notched composites by Boger [25] and Satyanarayana [26]. They predicted damage in composite plates notched in the center for different layups under tension. Both experimental results and simulation results showed that the crack path and failure initiation depends on fiber orientation.

In this section, the crack propagation path and dynamic fracture mode of unidirectional fiber-reinforced composites are studied using the proposed peridynamic model. The qualitative comparison of the peridynamic results with those from experiments are of interest.

Consider the compact tension test on a $100\text{ mm} \times 200\text{ mm}$ carbon/epoxy unidirectional composite plate with a 20 mm pre-notch at the center as shown in Fig. 5.16. The plate is loaded at the top and the bottom boundary by a uniform stress σ . The fiber orientation is α . The material

properties of the carbon/epoxy plate are shown in Table 5.6 [23].

Table 5.6 Material properties of graphite/epoxy

Longitudinal Young's modulus, E_1	329 GPa
Transverse Young's modulus, E_2	6 GPa
Poisson's ratio, ν_{12}	0.346
Shear modulus, G_{12}	4.4 GPa
Mode I intralamina fracture energy for longitudinal loading, G_{10}	15.49 kJ/m ²
Mode I intralamina fracture energy for transverse loading, G_{20}	0.168 kJ/m ²
Mass density, ρ	1630 kg/m ³

Fig. 5.17, Fig. 5.18, Fig. 5.19, Fig. 5.20 and Fig. 5.21 show the peridynamic simulation results for $\alpha = 0^\circ$, $\alpha = 30^\circ$, $\alpha = 45^\circ$, $\alpha = 60^\circ$ and $\alpha = 90^\circ$, respectively. In each figure, there are three contour plots with (a) vertical displacement of the plate, (b) strain energy density distribution and (c) local damage of the plate. Different color bars are associated with different plots with red indicating the highest value while blue the lowest value. Crack paths can be clearly seen from the strain energy density contour plot as there are always strain energy concentrations at the crack tips. The local damage is defined by Eqn. 2.12 and Eqn. 2.13, which show different levels of damage. A proper cutoff value can be defined to judge if there is a crack. In all cases, the crack propagates in the same direction as the fibers, which is consistent with the experimental observations from Wu [24]. The damage is due to the separation between matrix and fiber. There is no fiber breakage.

As expected, in the $\alpha = 0^\circ$ case, the crack propagates in the same direction as the pre-notch. This also matches with the computational and experimental results in Section 5.6. In the smaller angle case $\alpha = 30^\circ$, aside from the major crack, which propagates in 30° , there is matrix shattering at the sides of the plate in 0° direction. The matrix shattering happens before the crack starts to propagate as shown in Fig. 5.22. It starts at the lateral of the plate and propagates to the interior of the plate. From Fig. 5.18 (c), the matrix shattering is not as severe as the major crack since only 20% of the bonds are broken. However, in the major crack, more than 70% of the bonds are broken. This is why matrix shattering is only a material softening and may not be seen from experimental observation as reported in [24]. For the $\alpha = 90^\circ$ case, the composite plate fails due to splitting caused by shear stress in the matrix. It which matches with the findings of Boger [25].

5.8 Conclusions

A peridynamic orthotropic material model based on the beam model is proposed in this chapter. There are four independent material parameters in this model and it matches with the four material properties for two-dimensional orthotropic materials. The bond material properties depend on these four material parameters and the angle between the bond orientation and fiber orientation. This results in the continuity of the bond stiffness function with no need of remeshing for different fiber orientations. This model is verified by a static tensile test and a vibration problem of a laminated beam.

Dynamic damage propagation problems in composite materials can be greatly benefited from peridynamics. The prediction of damage initiation and crack propagation of composite materials is

complex using traditional methods, such as finite element analysis, due to its anisotropy. As investigated in peridynamic simulations, there is no need of tracking each crack propagation, finding different damage modes and applying different damage rules. Damage happens automatically. A single edge notch test is simulated in this chapter and the results match with the experimental results. Crack path and failure initiation of a center notch plate is predicted successfully by peridynamics when comparing with the experimental results.

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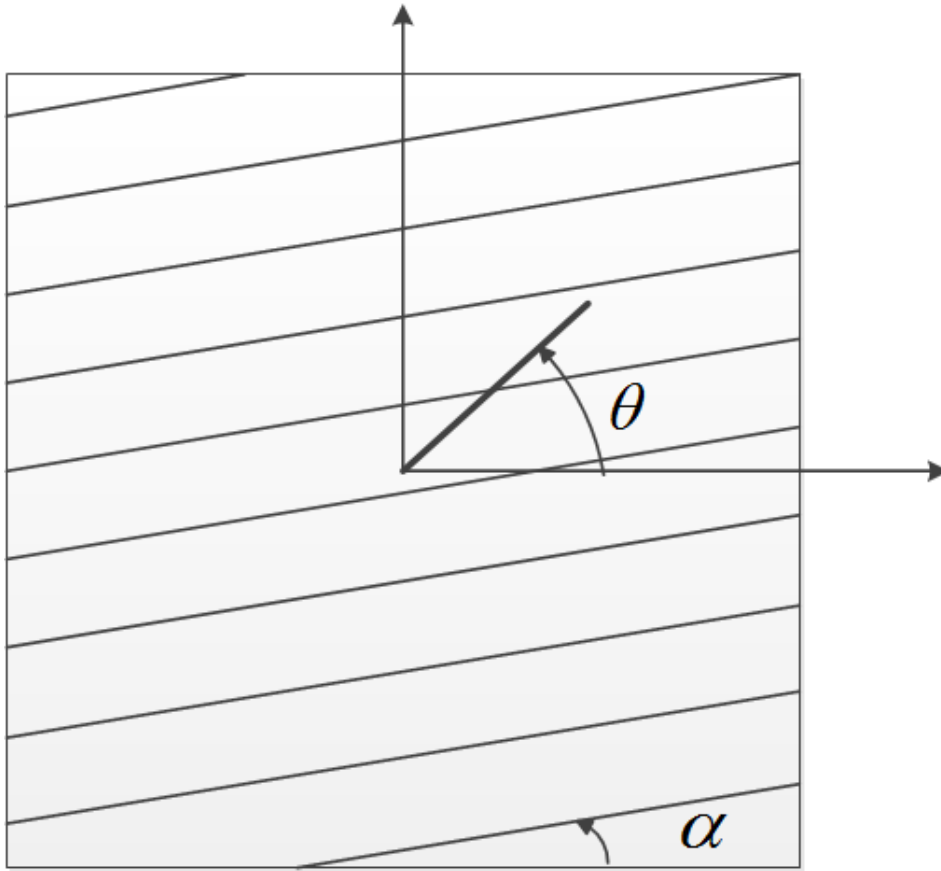


Figure 5.1 A composite plate with fiber in α° direction and a bond in θ° direction.

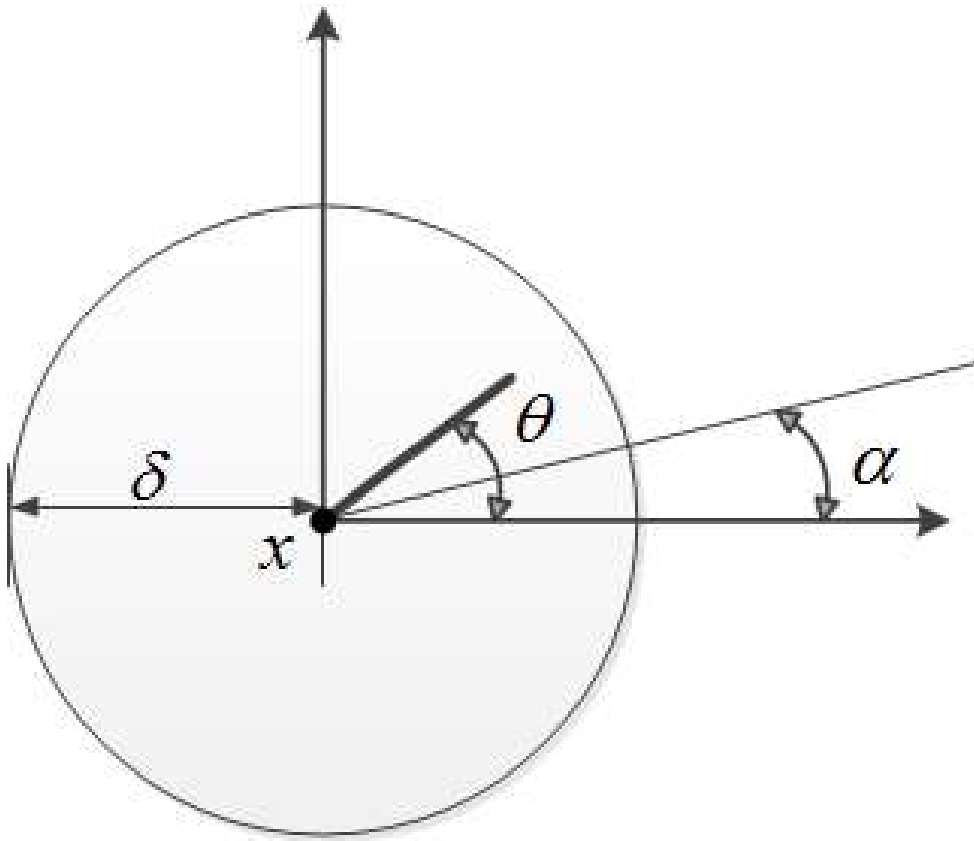


Figure 5.2 Coordinate system for calculating strain energy density at point x .

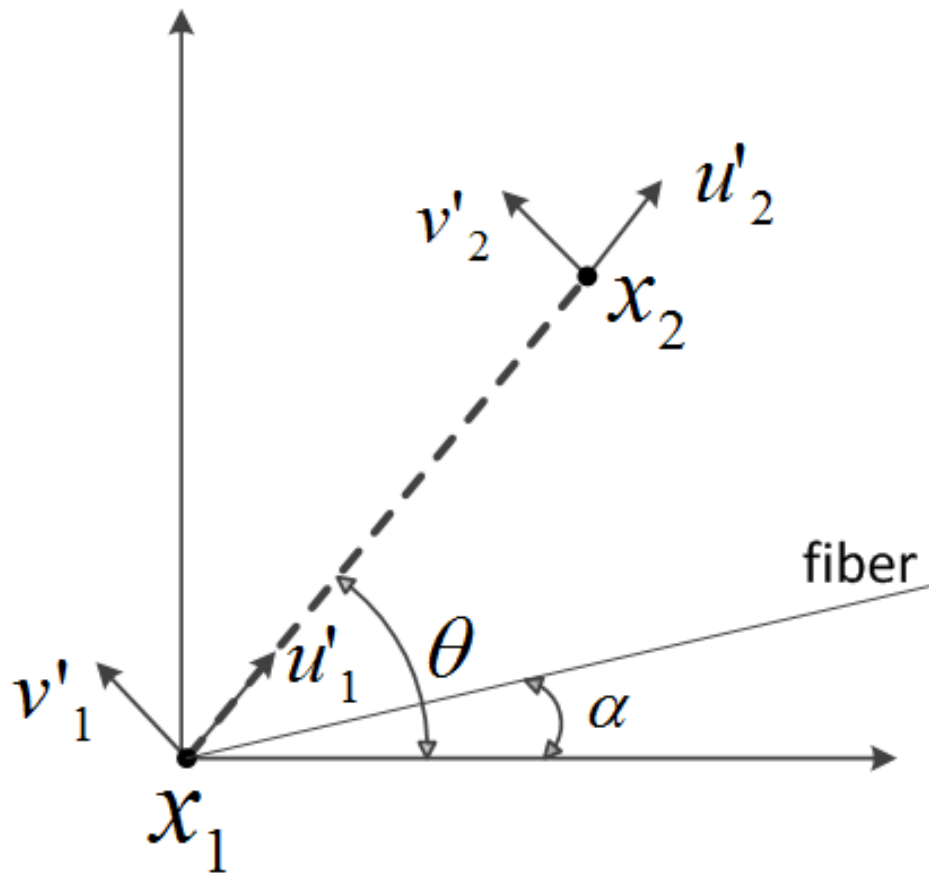


Figure 5.3 Beam model for orthotropic materials.

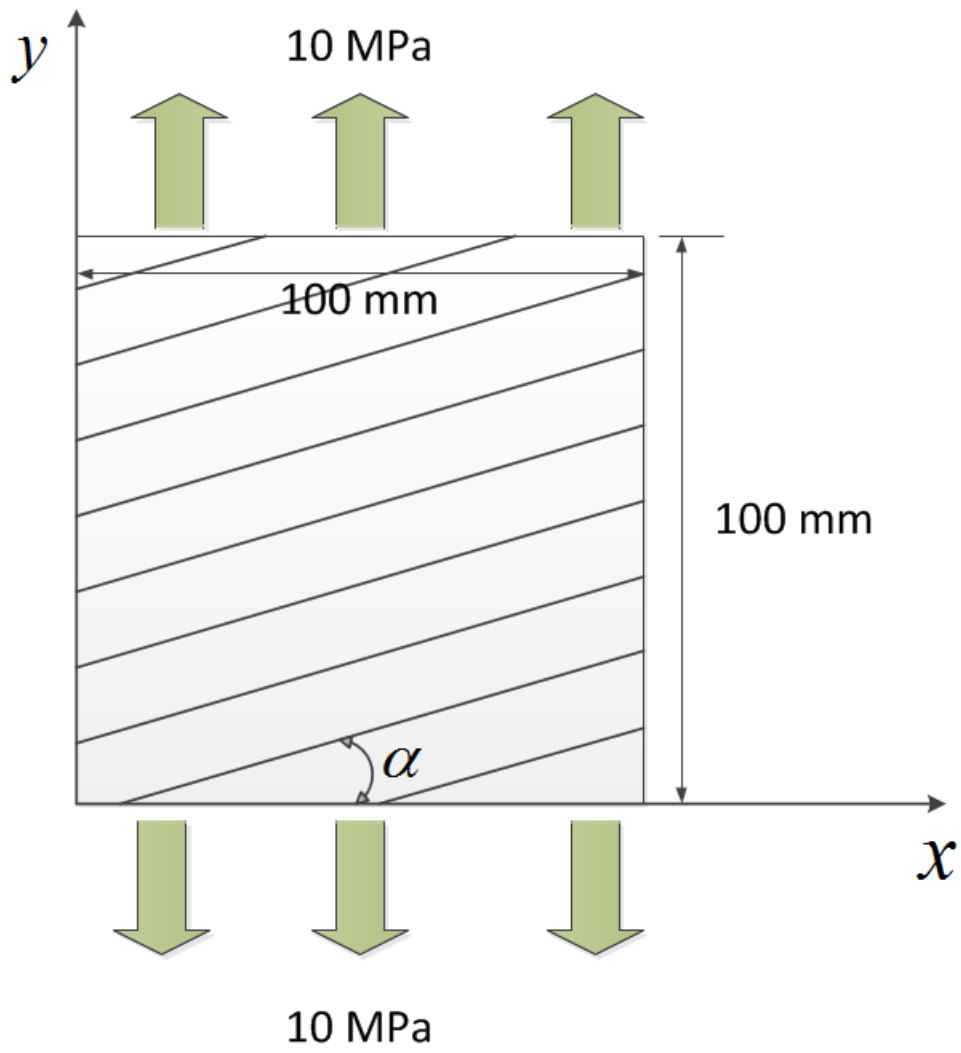


Figure 5.4 Pulling test in a composite laminate.

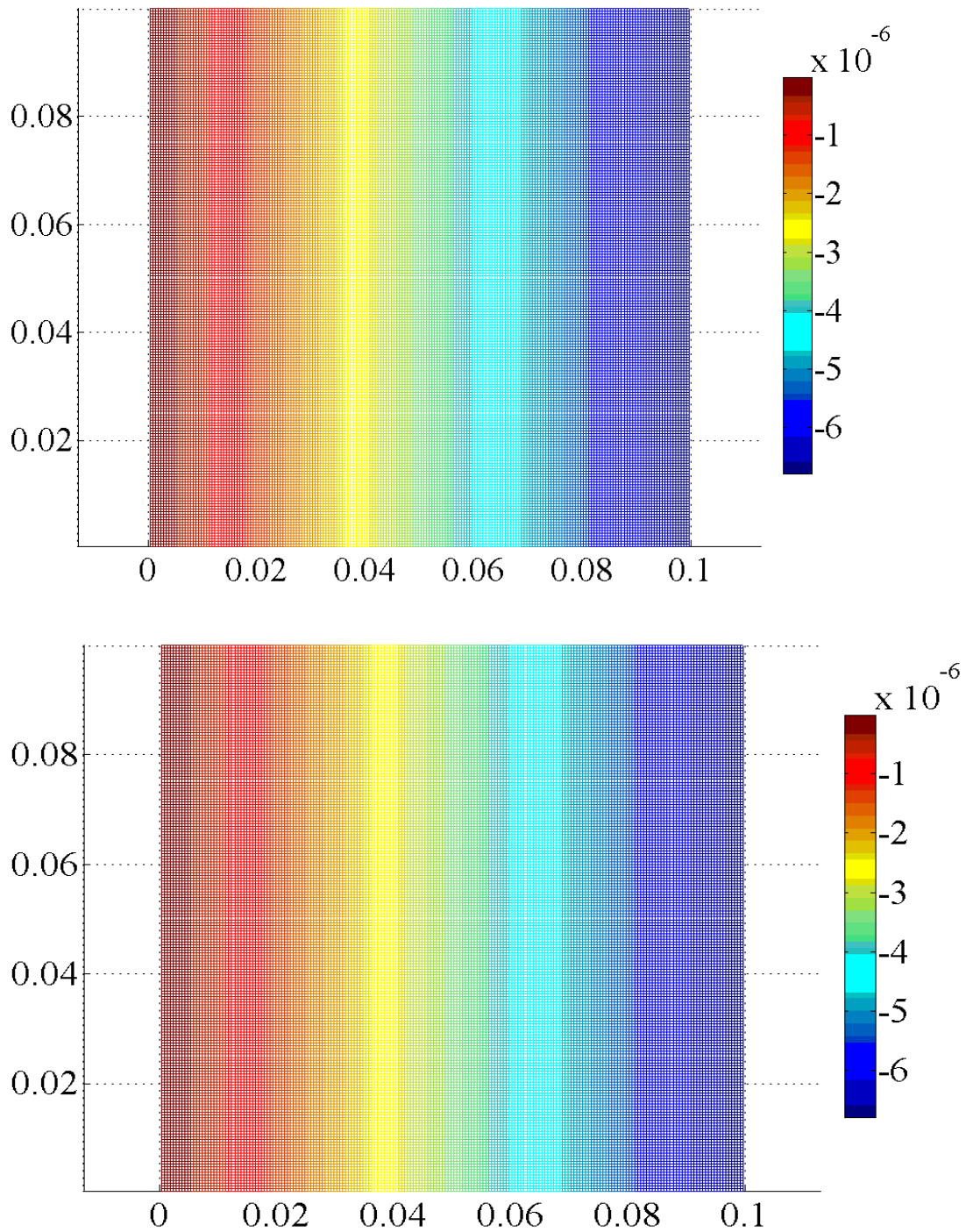


Figure 5.5 Comparison of u_x of 0° laminate calculated from peridynamics (top) and composite theory (bottom).

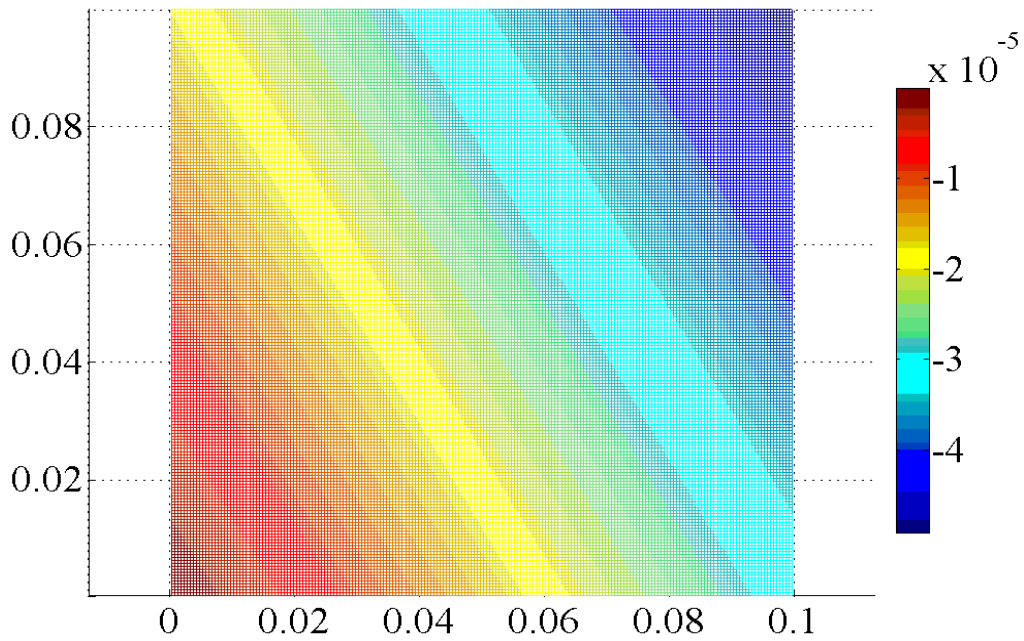
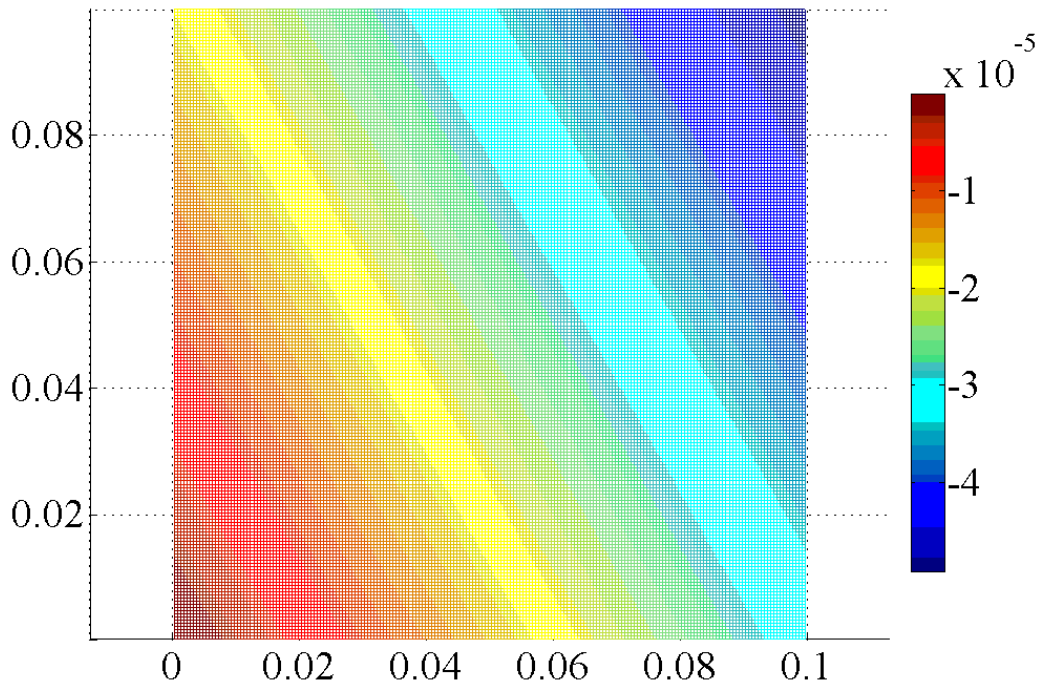


Figure 5.6 Comparison of u_x of 45° laminate calculated from peridynamics (top) and the composite theory (bottom).

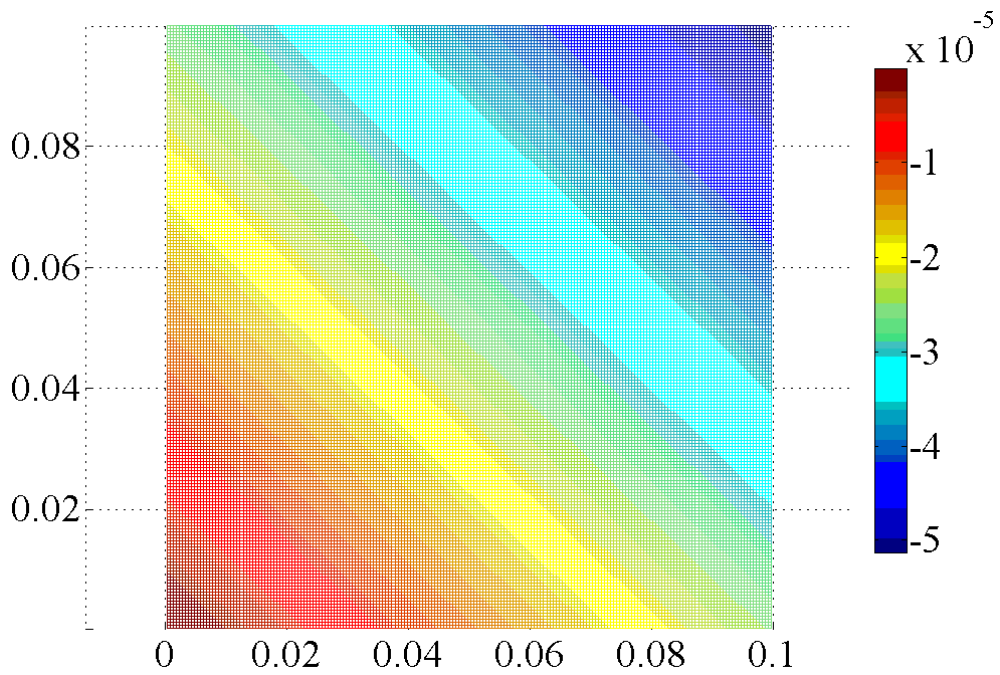
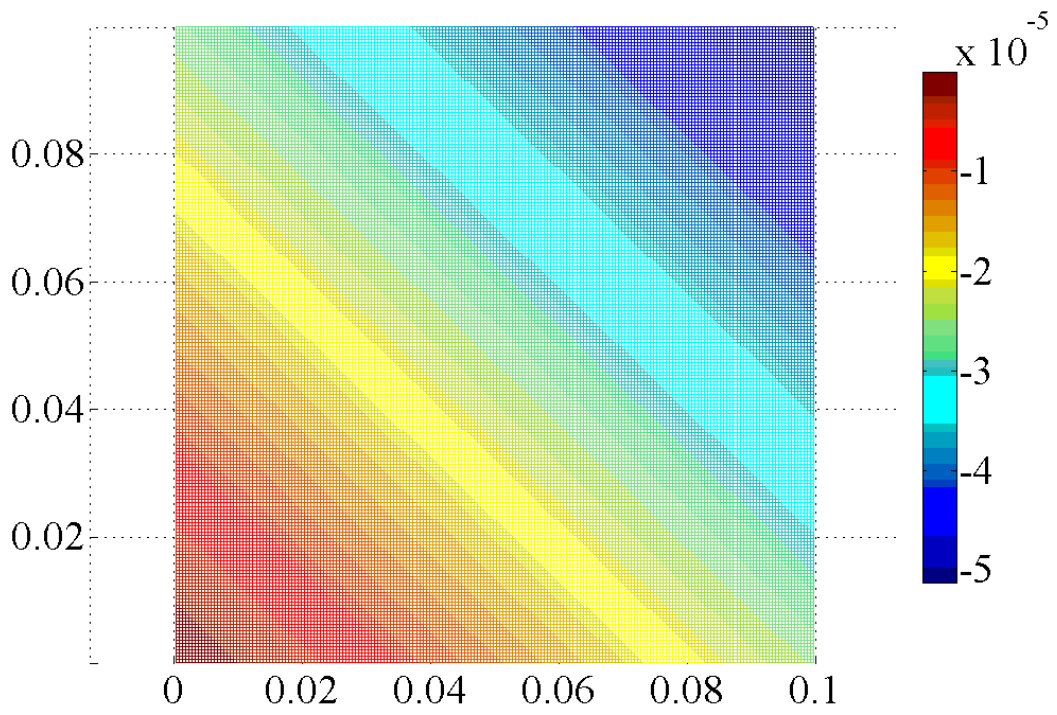


Figure 5.7 Comparison of u_x of 60° laminate calculated from peridynamics (top) and the composite theory (bottom).

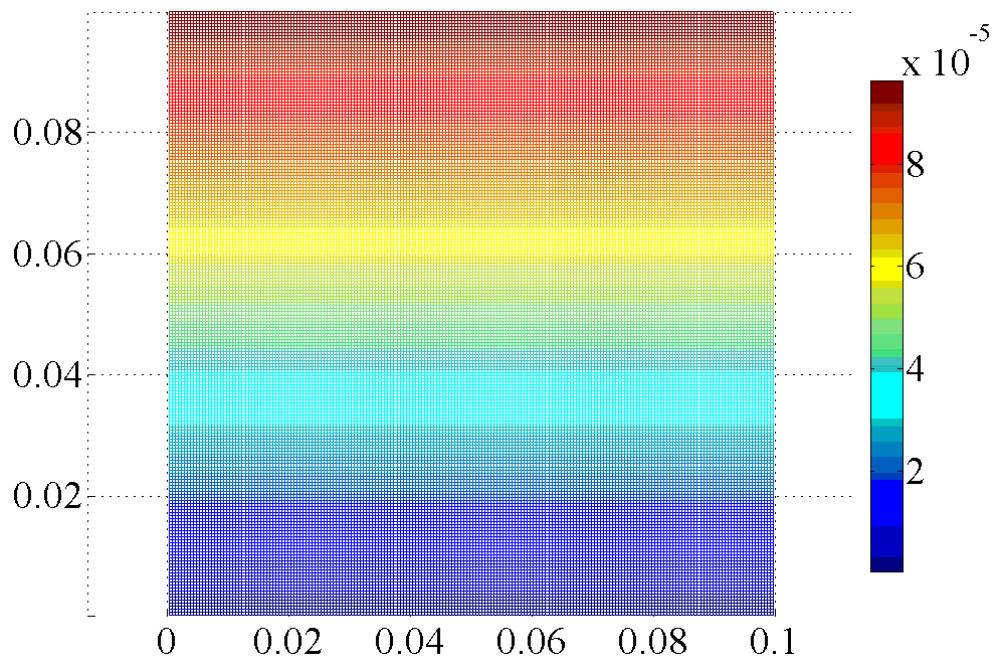
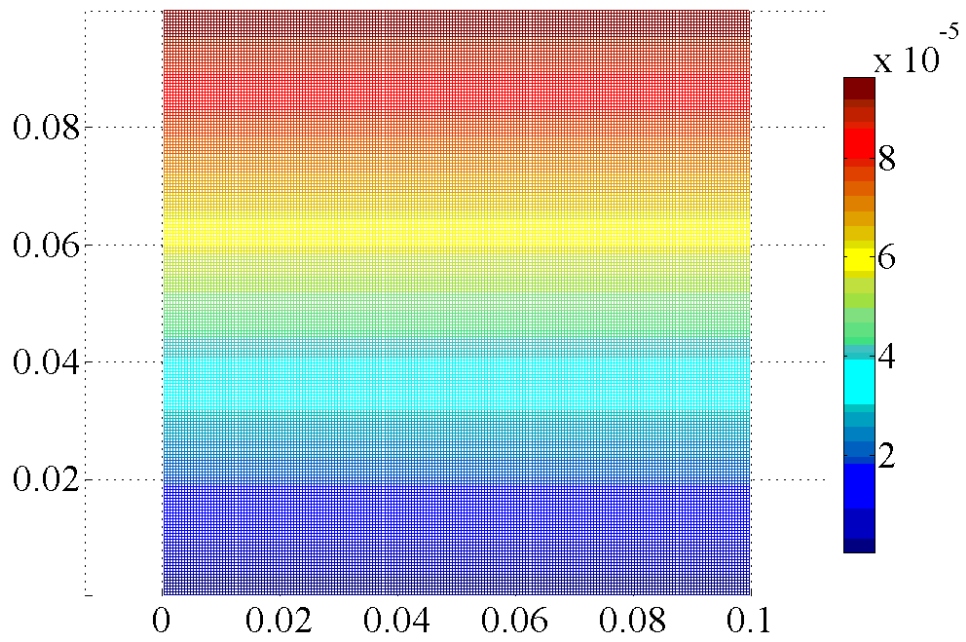


Figure 5.8 Comparison of u_y of 0° laminate calculated from peridynamics (top) and the composite theory (bottom).

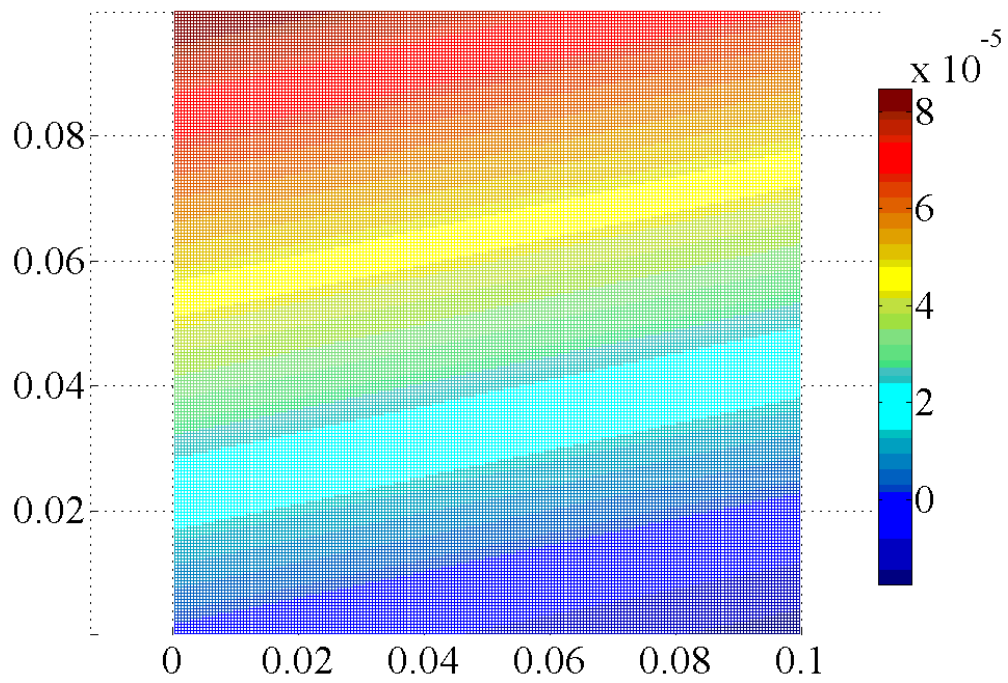
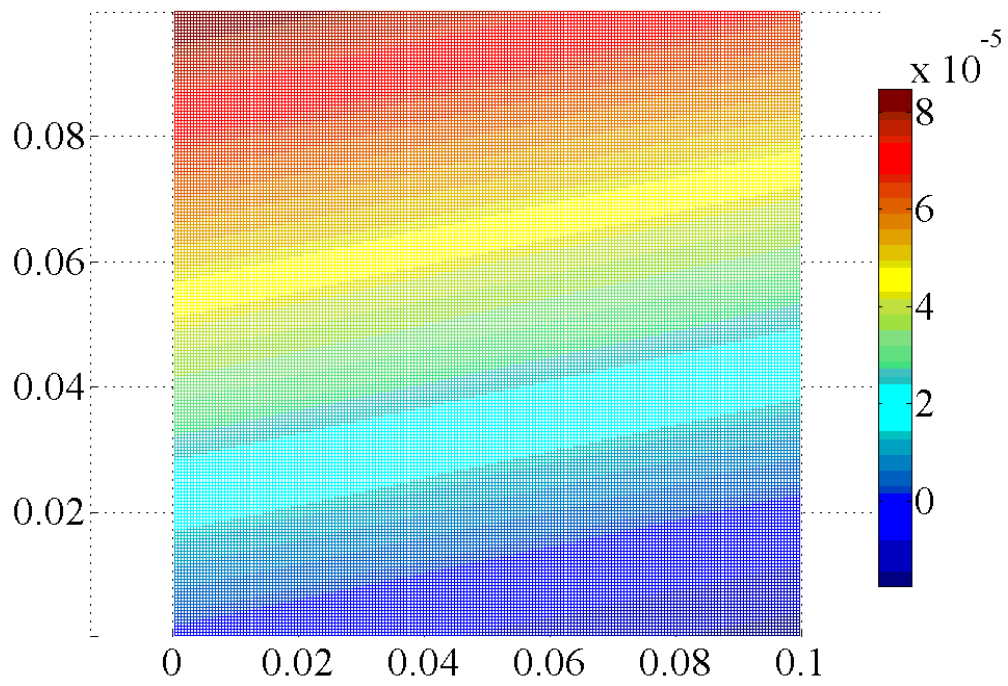


Figure 5.9 Comparison of u_y of 45° laminate calculated from peridynamics (top) and the composite theory (bottom).

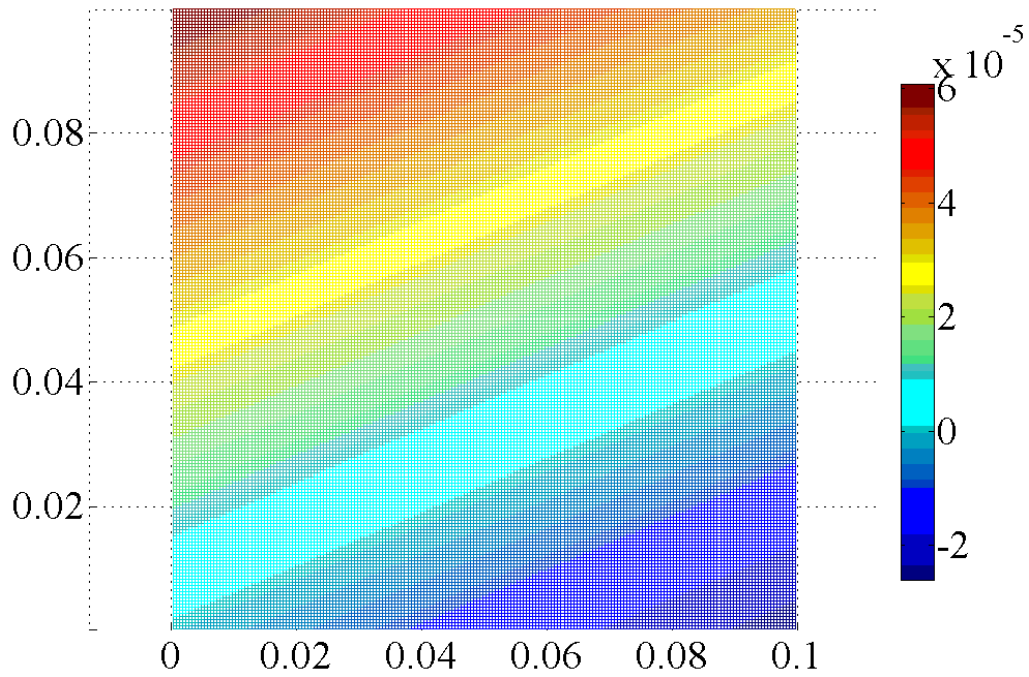
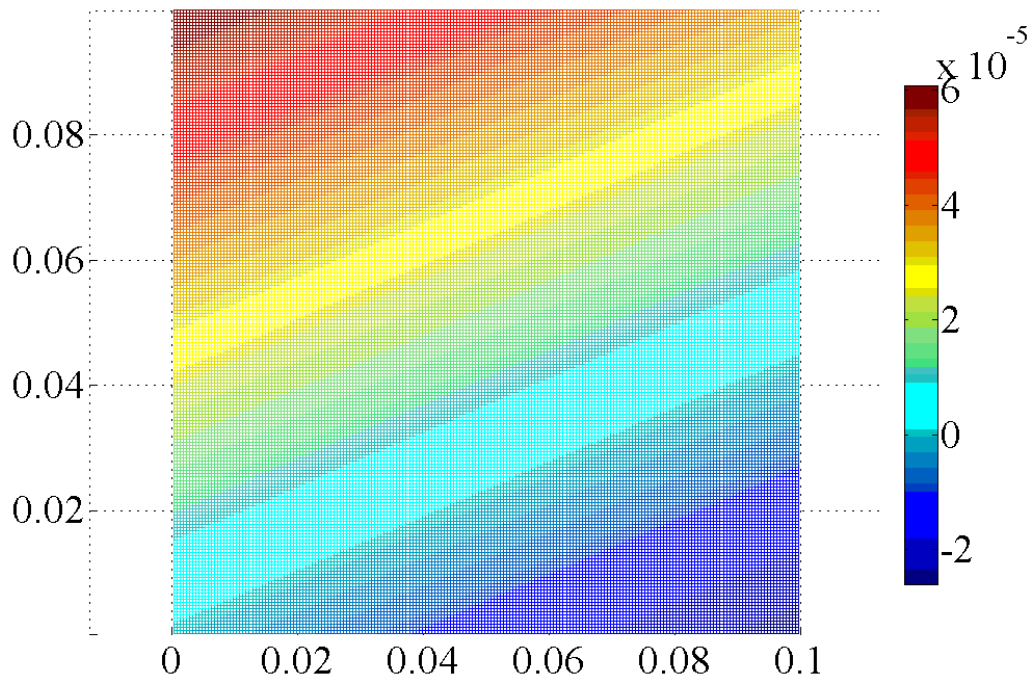


Figure 5.10 Comparison of u_y of 60° laminate calculated from peridynamics (top) and the composite theory (bottom).

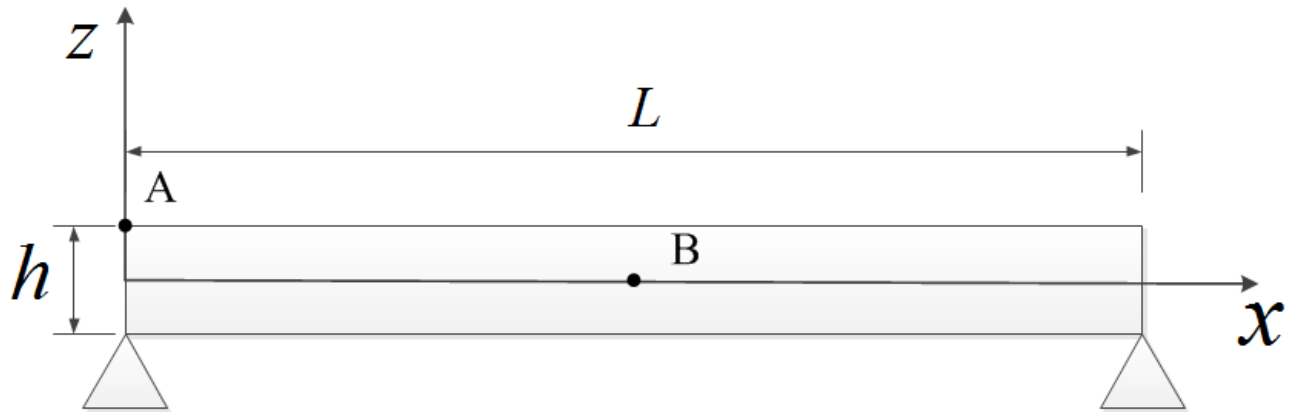


Figure 5.11 Free vibration of a laminated beam with fibers in x direction.

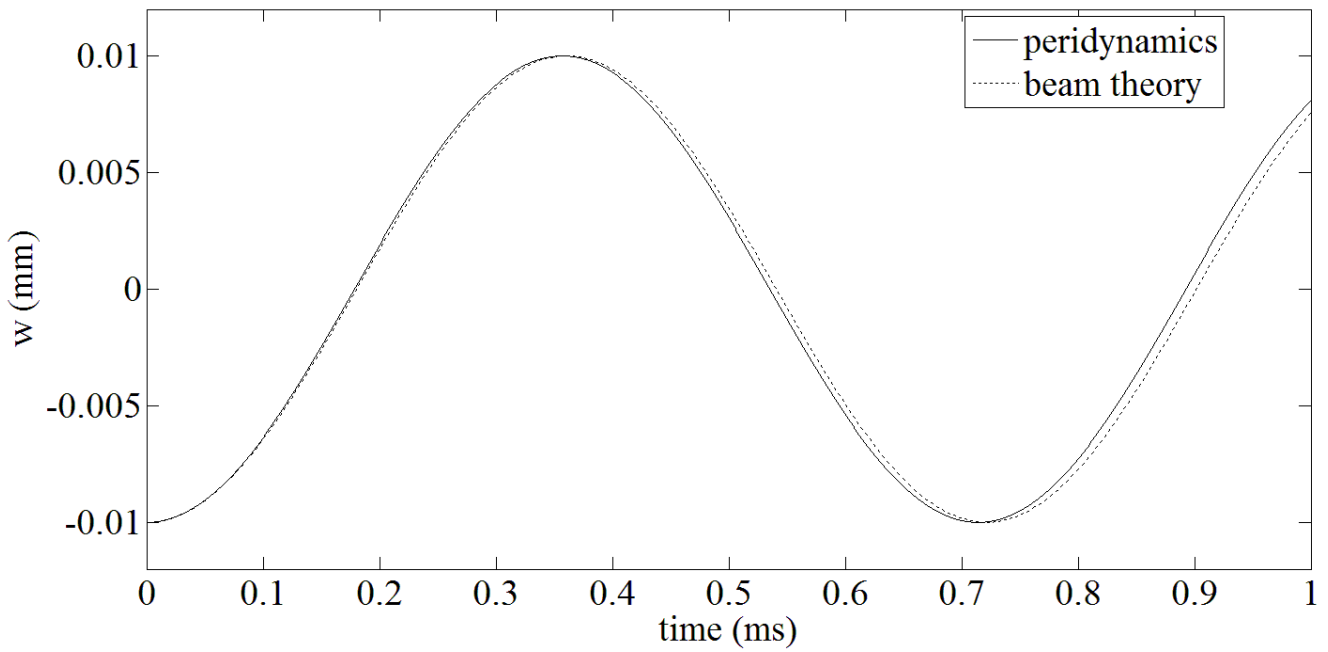
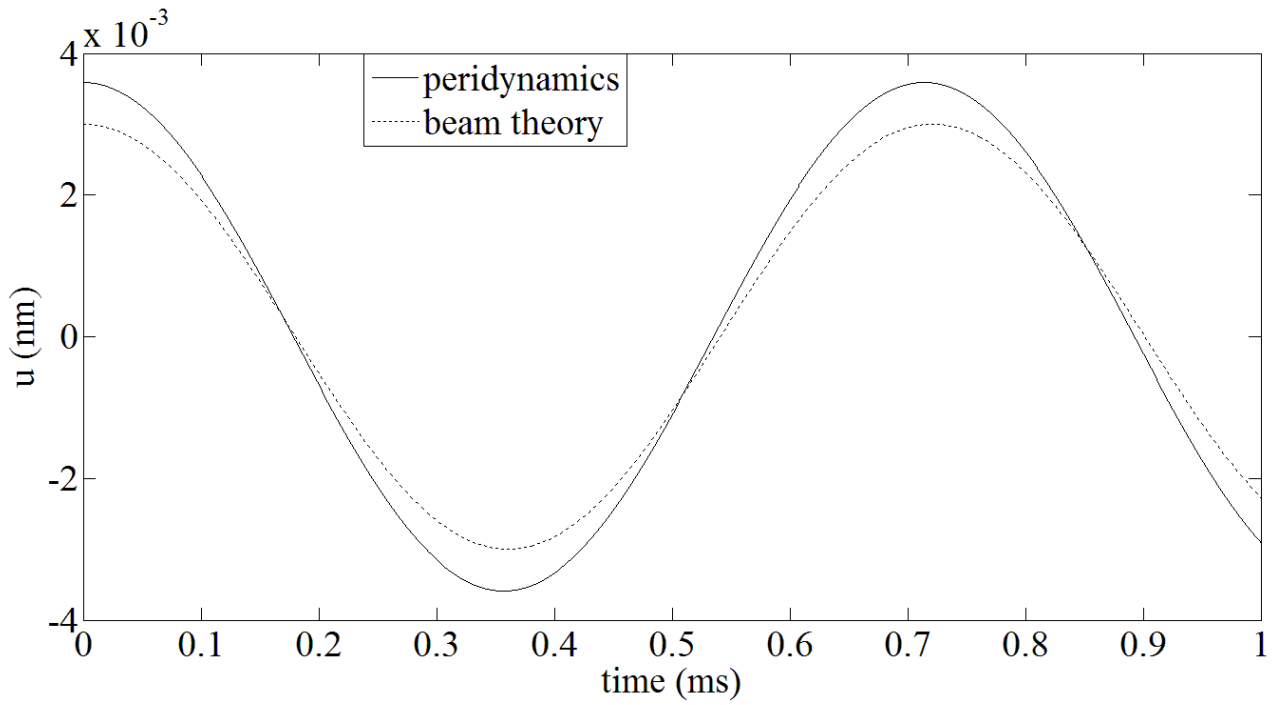


Figure 5.12 Peridynamic results compared with composite beam theory results with $a = 5$.

Top: horizontal displacement u of point A. Bottom: vertical displacement w of point B.

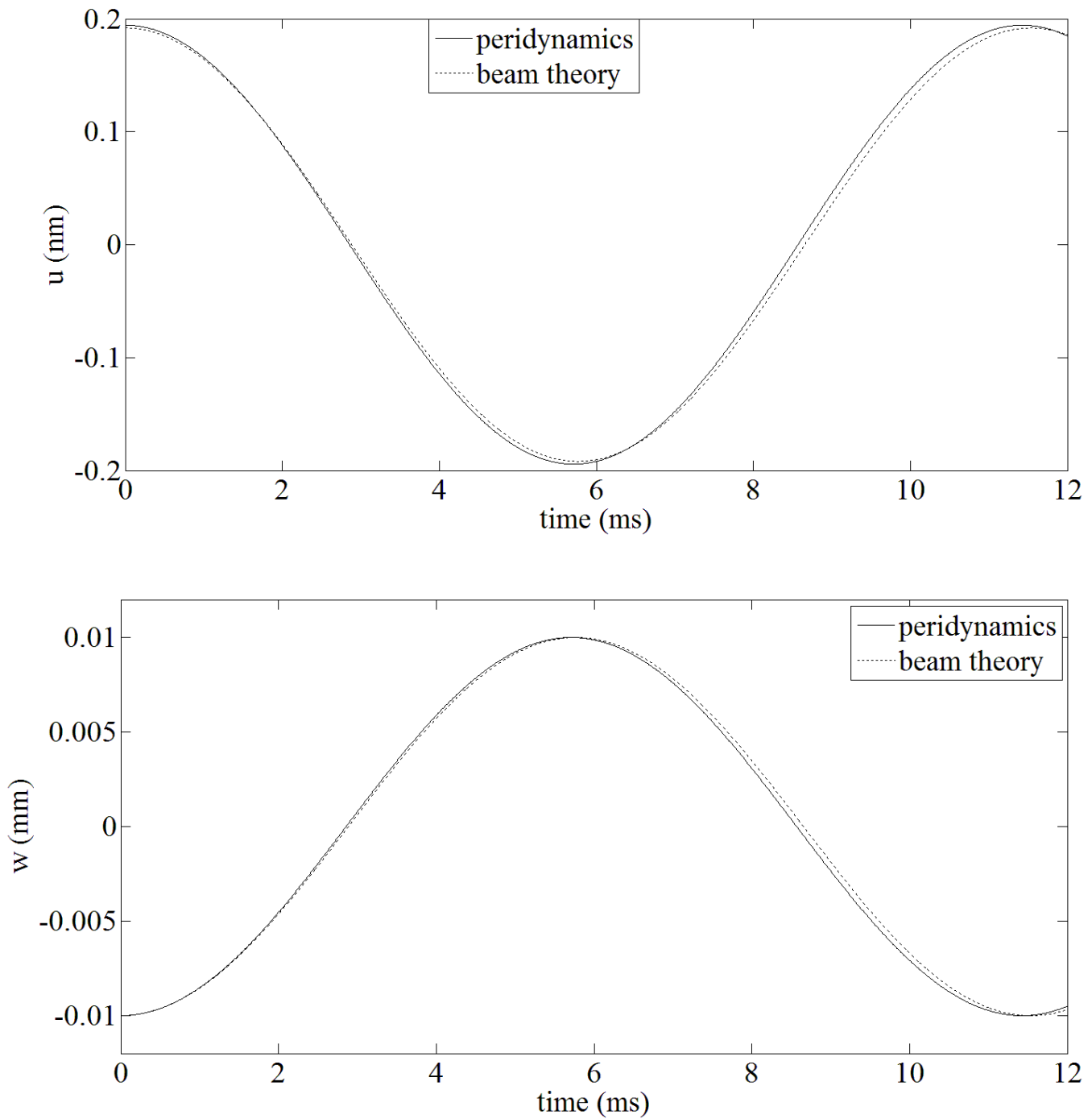


Figure 5.13 Peridynamic results compared with composite beam theory results with $a = 20$.

Top: horizontal displacement u of point A. Bottom: vertical displacement w of point B.

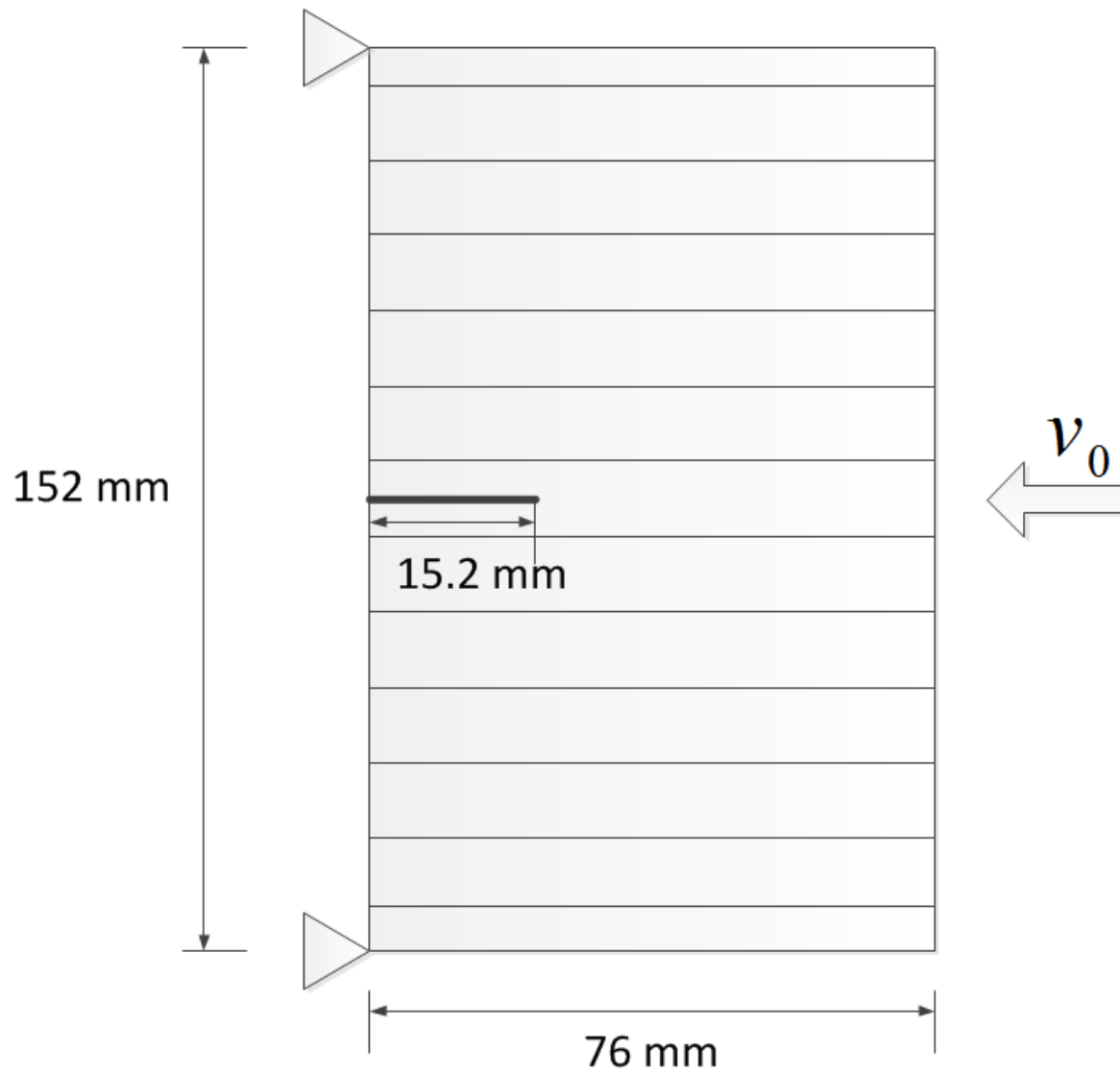


Figure 5.14 An unidirectional composite plate with single edge notch under three point bending.

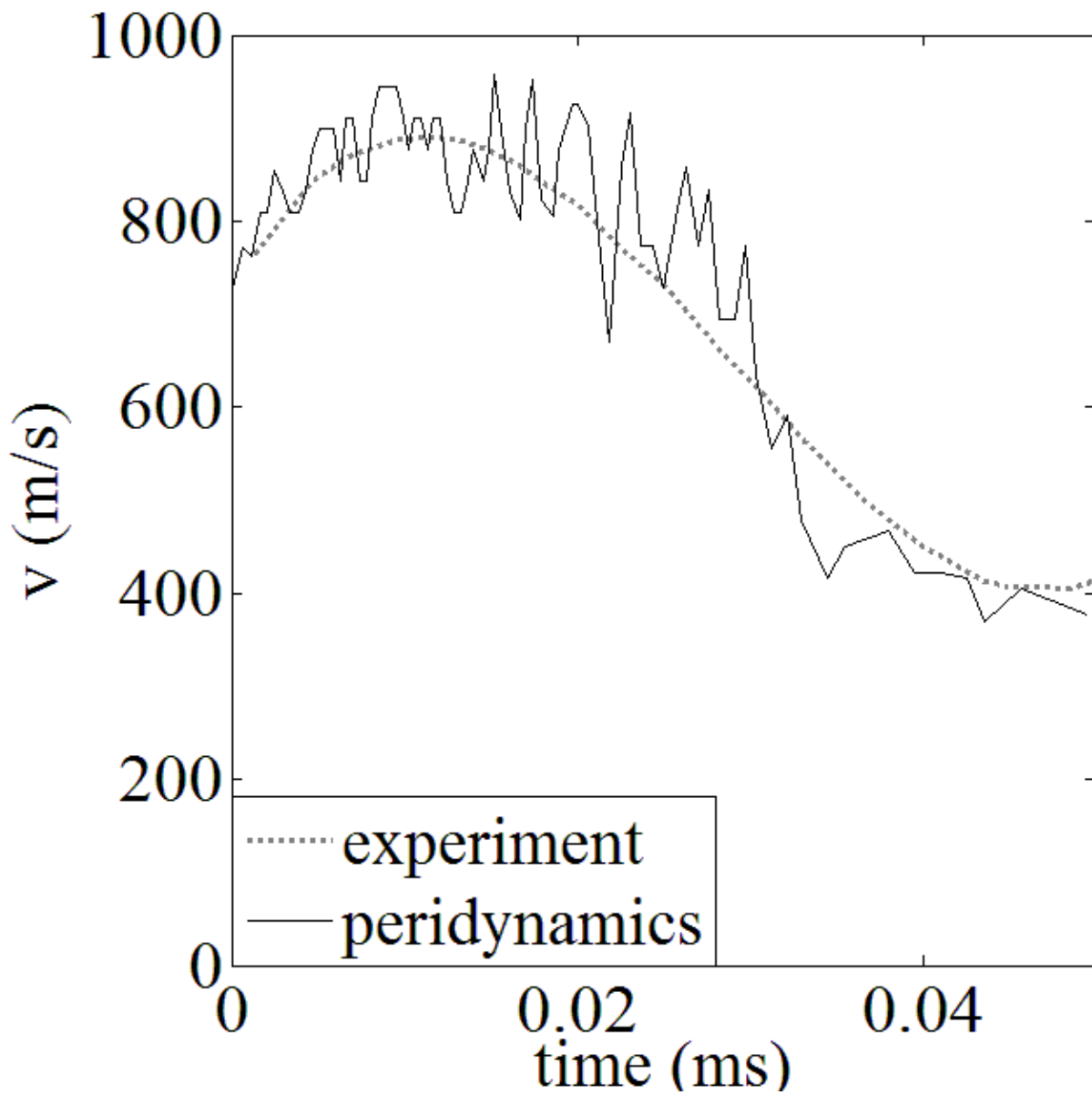


Figure 5.15 Comparison of crack propagation velocity between peridynamics and experiment.

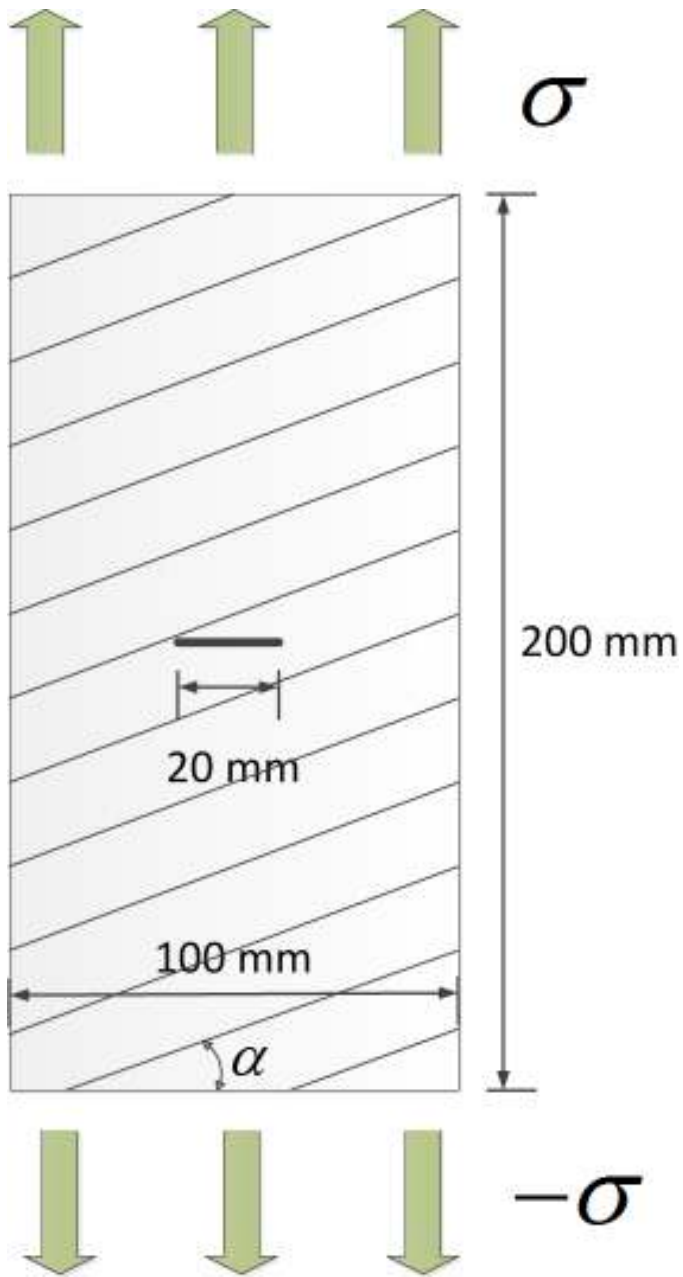


Figure 5.16 Compact tension test for a unidirectional composite plate.

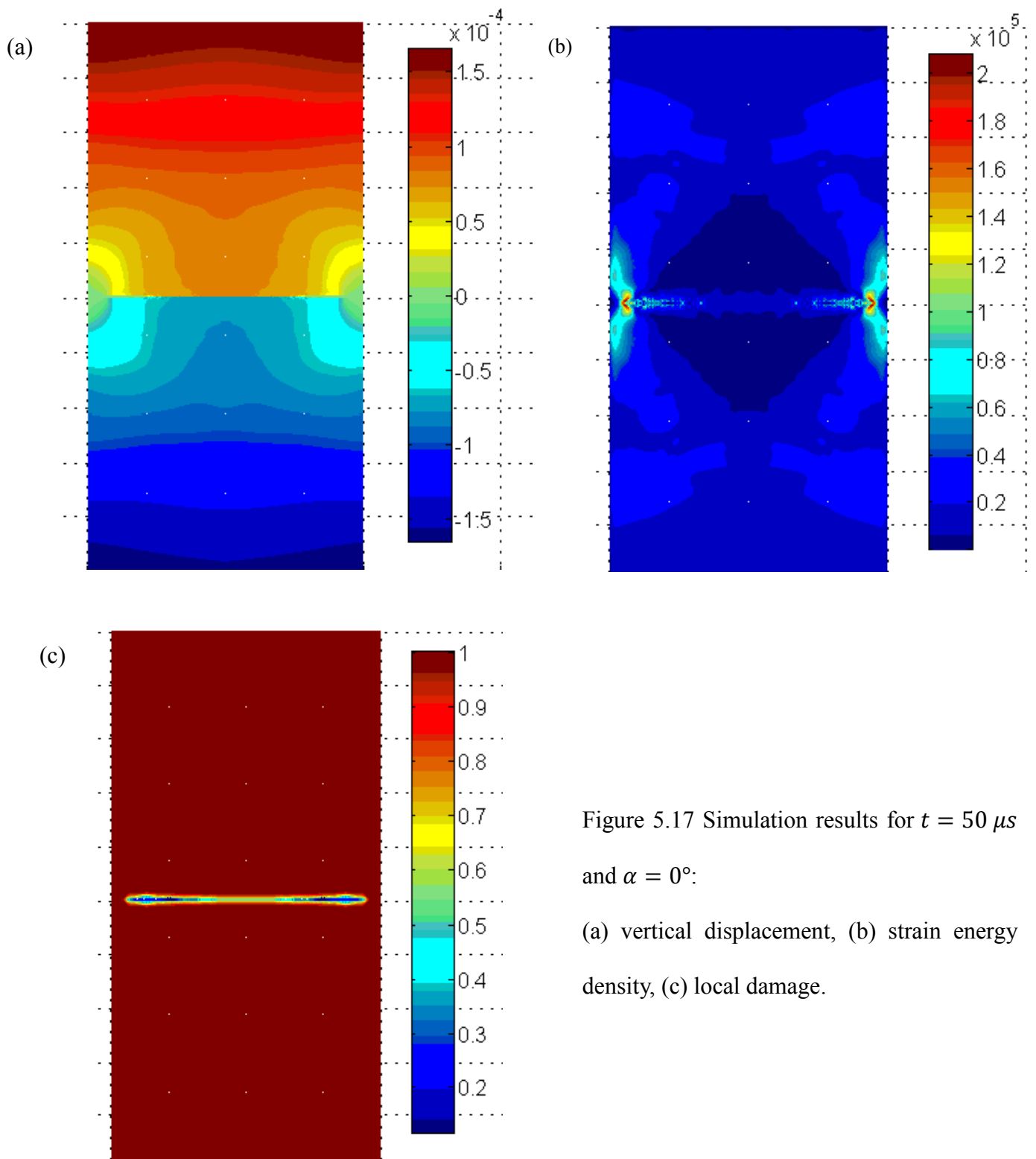


Figure 5.17 Simulation results for $t = 50 \mu s$ and $\alpha = 0^\circ$:

(a) vertical displacement, (b) strain energy density, (c) local damage.

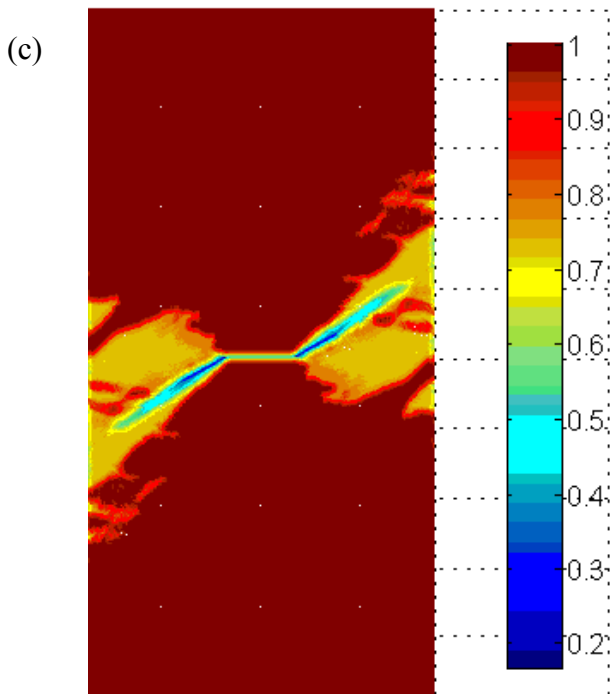
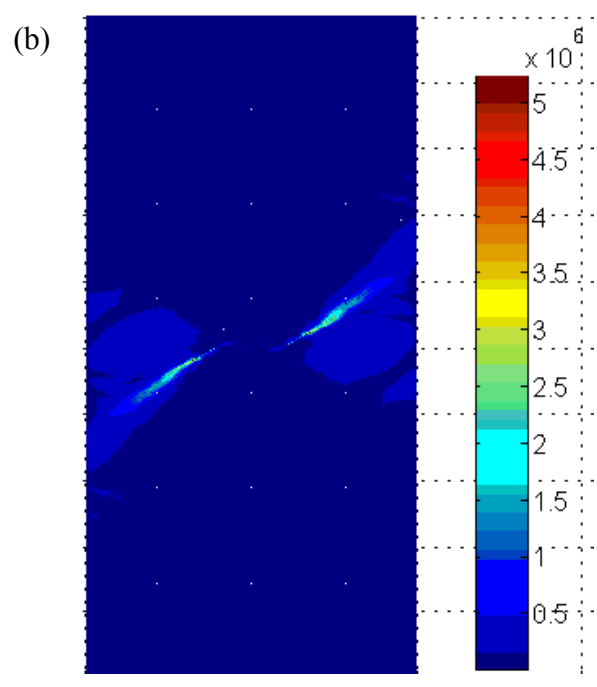
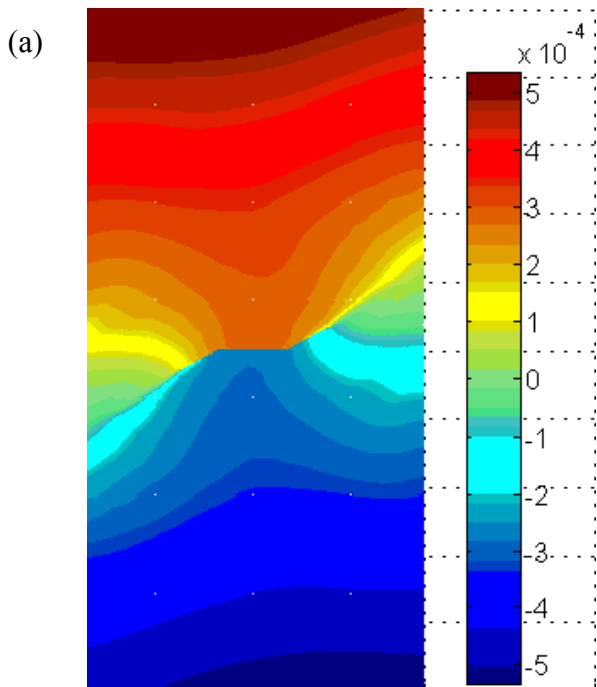


Figure 5.18 Simulation results for $t = 70 \mu s$ and $\alpha = 30^\circ$:

(a) vertical displacement, (b) strain energy density, (c) local damage.

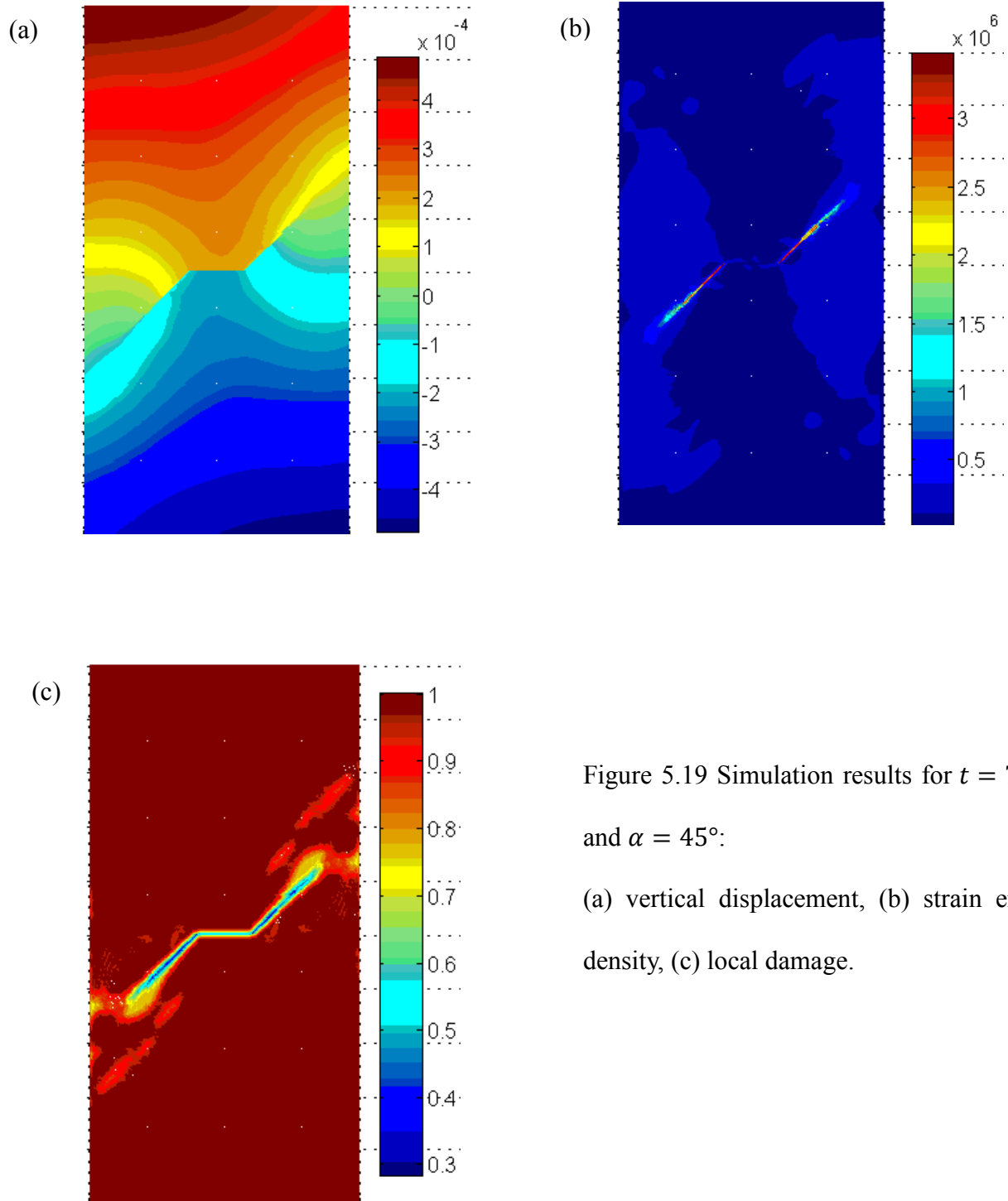


Figure 5.19 Simulation results for $t = 70 \mu s$ and $\alpha = 45^\circ$:
 (a) vertical displacement, (b) strain energy density, (c) local damage.

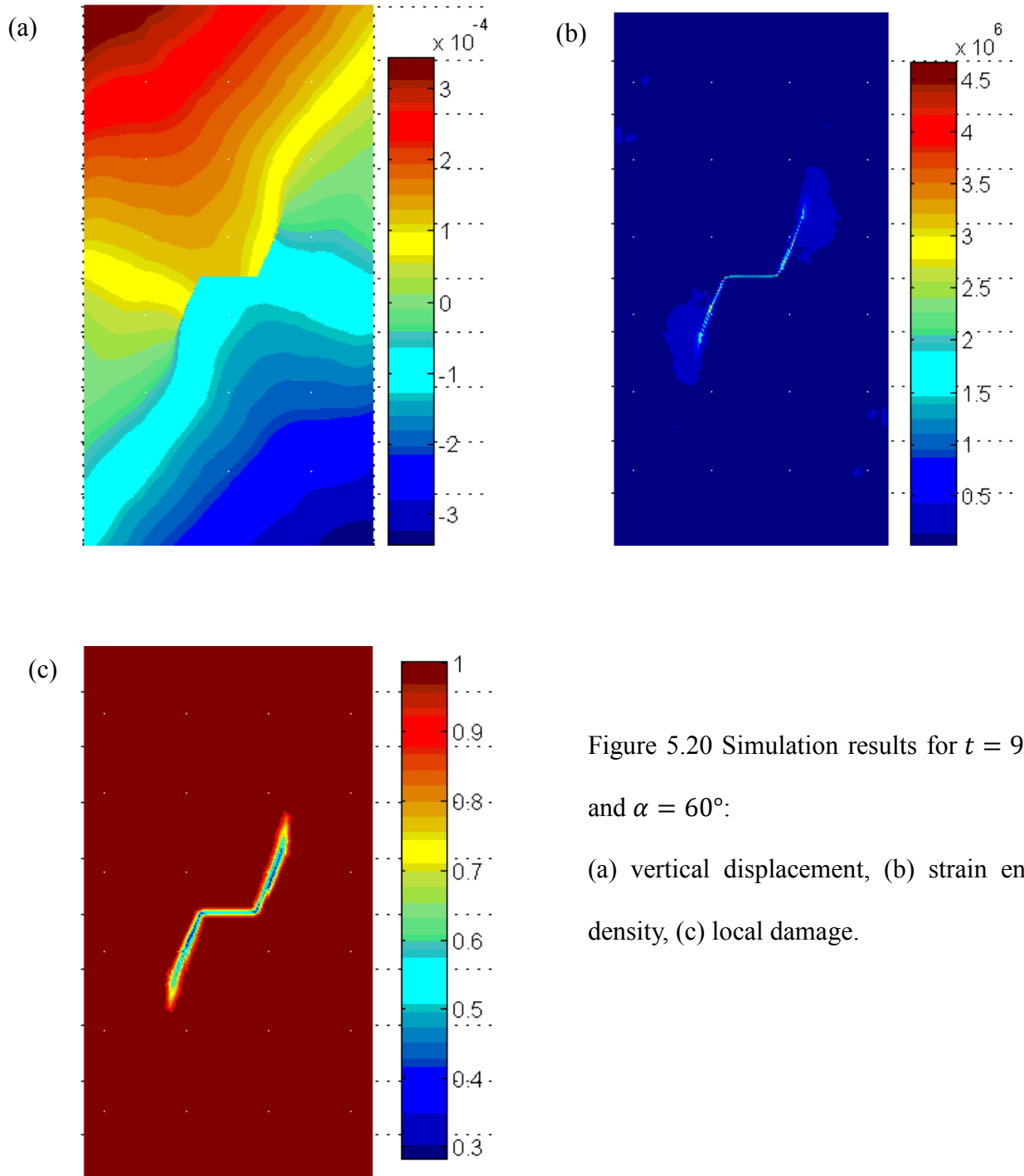


Figure 5.20 Simulation results for $t = 90 \mu\text{s}$ and $\alpha = 60^\circ$:
 (a) vertical displacement, (b) strain energy density, (c) local damage.

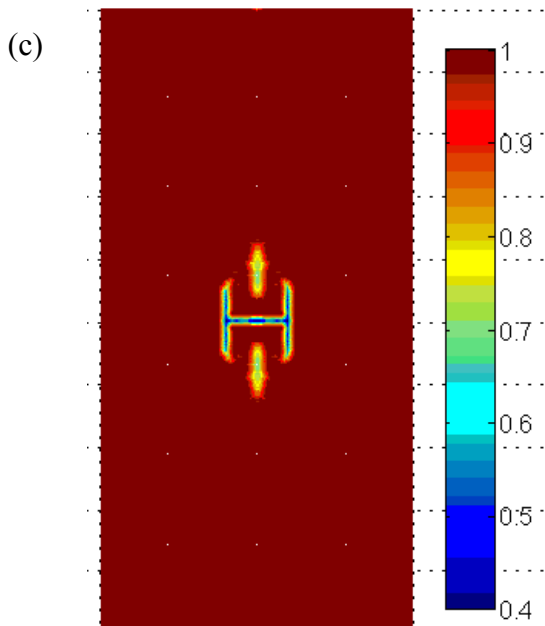
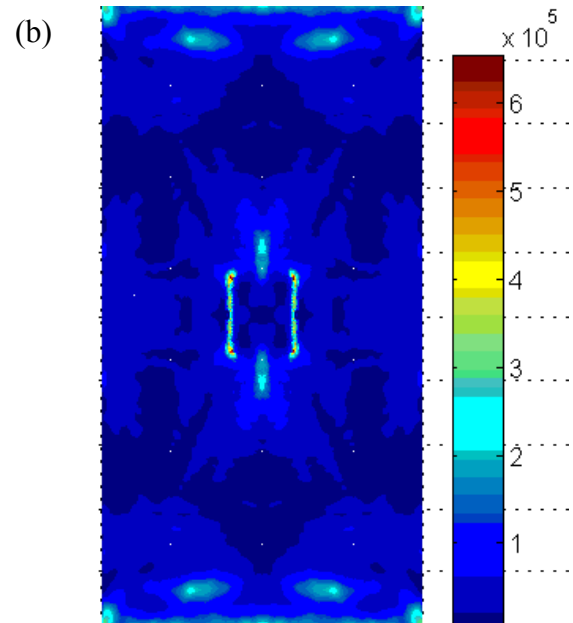
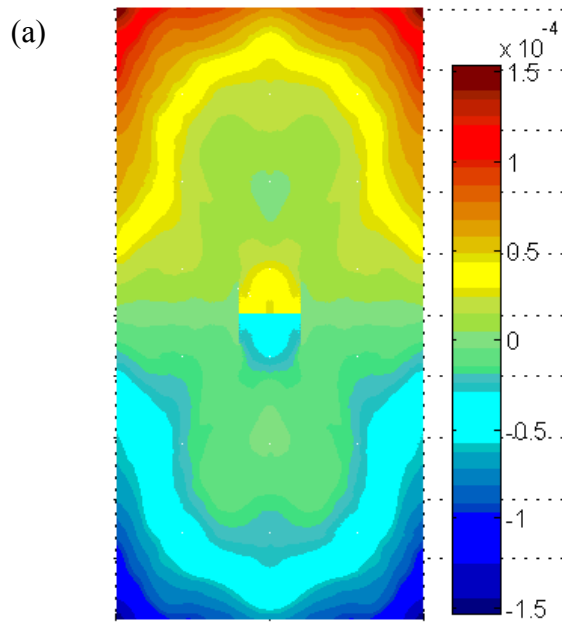


Figure 5.21 Simulation results for $t = 100.5 \mu s$ and $\alpha = 90^\circ$:

(a) vertical displacement, (b) strain energy density, (c) local damage.

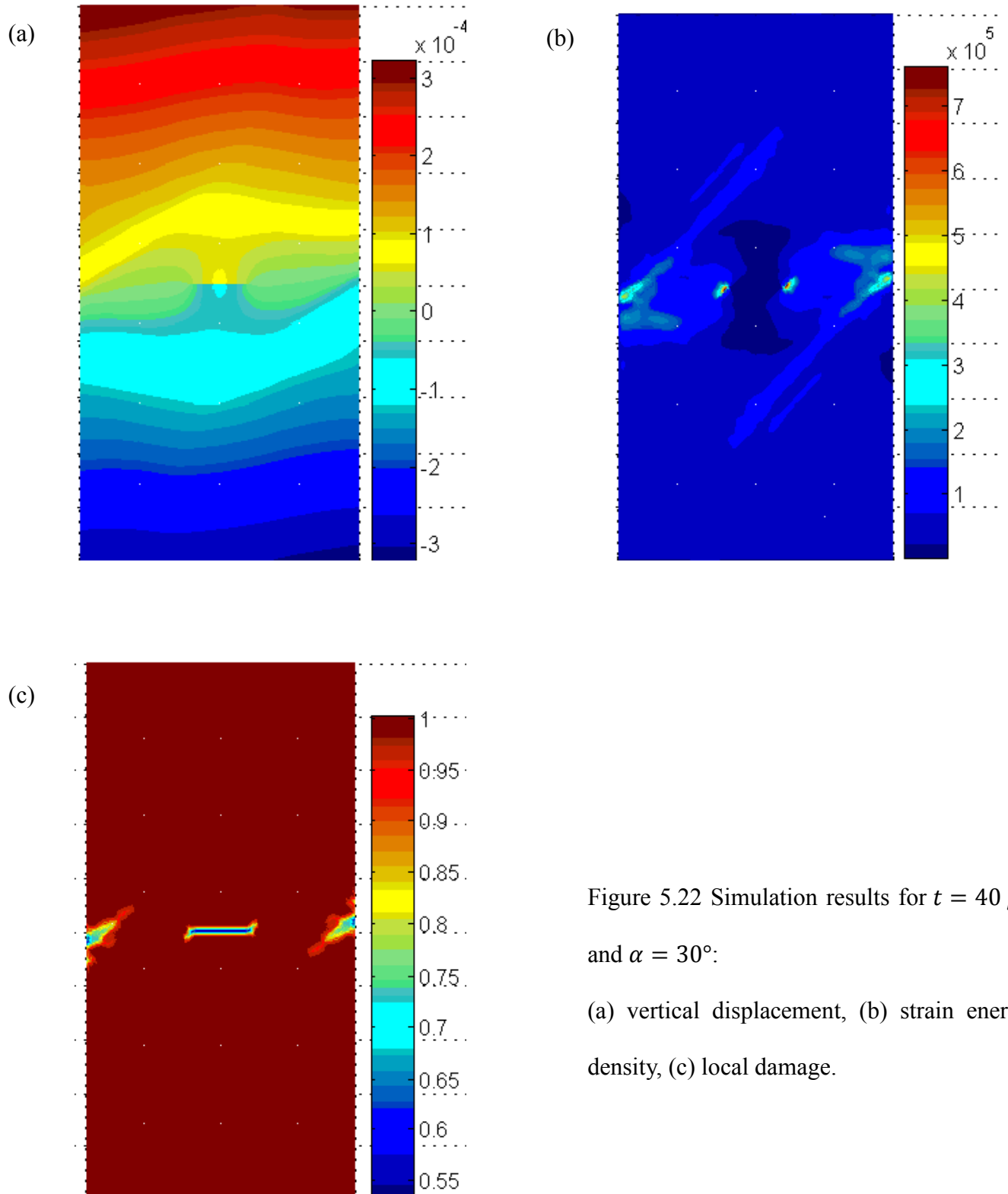


Figure 5.22 Simulation results for $t = 40 \mu s$ and $\alpha = 30^\circ$:

(a) vertical displacement, (b) strain energy density, (c) local damage.