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Formation of MOS Gates by
Rapid Thermal/Microwave Remote Plasma Multiprocessing

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Abstract

A novel cold-wall single-wafer lamp-heated Rapid Thermal/Microwave Remote Plasma Multiprocessing (RTMRPM) reactor has been developed for multilayer in-situ growth and deposition of dielectrics, silicon, and metals. This equipment is the result of an attempt to enhance semiconductor processing equipment versatility, to improve process reproducibility and uniformity, to increase growth and deposition rates at reduced processing temperatures, and to achieve in situ multiprocessing in conjunction with real-time process monitoring and automation. For high-performance MOS VLSI applications, a variety of selective and nonselective tungsten deposition processes were investigated in this work. The tungsten gate MOS devices fabricated using the remote plasma multiprocessing techniques exhibited negligible plasma damage and near-ideal electrical characteristics. The flexibility of the reactor allows optimization of each process step yet allows multiprocessing.
Introduction

Future technological advancements in integrated electronics will require development of flexible custom fabrication technology for custom VLSI systems. Low temperatures and short times are essential requirements of future VLSI processing and the use of plasma processing in conjunction with single-wafer lamp heating in a flexible environment is a major step to realize this goal. As demonstrated in this work, the combination of single-wafer rapid thermal processing and microwave remote plasma can provide a powerful multipurpose reactor for VLSI device fabrication. *In-situ* multiprocessing reduces contamination, enhances circuit yield, and makes the formation of numerous new device structures feasible. The multipurpose capabilities of rapid thermal processing (RTP) have already been established through its application to silicon epitaxial growth [1], and *in-situ* fabrication of silicon gate MOS capacitors [2].

Reproducible growth of thin dielectrics in hot-wall furnaces is difficult due to long ambient and temperature transient times and constant furnace temperatures. Since furnaces are designed for multiwafer processing, extensive *in-situ* real-time measurements are difficult to perform. Rapid thermal oxidation and nitridation (RTO and RTN) of silicon in oxygen and ammonia ambients has already been recognized as an attractive technique for the growth of silicon nitride, silicon dioxide, nitrided oxides, oxidized nitrides, and application-specific (composition-tailored) insulators [3]. We have also demonstrated the feasibility of low-temperature nitridation of silicon in nitrogen plasma remotely generated by microwave discharge [4]. Among other growth and deposition processes, LPCVD of tungsten has emerged as a viable technology for VLSI applications such as MOS gate electrodes [5], low resistivity contacts and contact barriers, multilevel interconnections, and reduction of source/drain parasitic resistance. The conventional hot-wall LPCVD furnaces are not appropriate for reproducible high-rate tungsten deposition and nonselective formation of tungsten on insulators. This paper presents results on application of this
multiprocessing technique to deposit tungsten layers for semiconductor integrated circuit technologies.

**Multiprocessing Reactor Design**

Based on our experiences with remote microwave plasma nitridation and the rapid thermal dielectric growth and anneal processes, we have developed a novel cold-wall single-wafer *Rapid Thermal/Microwave Remote Plasma Multiprocessing* (RTMRPM) reactor for *in-situ* growth and deposition of dielectrics, silicon, and metals. The reactor design is so comprehensive that several processing steps can be sequentially done in situ, at the same time the reactor is highly flexible allowing optimization of each processing step. This reactor is expected to enhance equipment versatility, to improve process reproducibility and uniformity, to increase growth and deposition rates, and to achieve *in-situ* semiconductor multiprocessing. Figure 1 shows the schematic of the simplified prototype design employed to obtain the preliminary results presented in this letter. The water-cooled stainless steel chamber provides various ports for gas injection, optical heating of the wafer, vacuum pumping, and *in-situ* process monitoring. The wafer sits on low thermal mass quartz pins facing the end cone of a discharge tube and is heated on the other side by arrays of tungsten-halogen lamps (multiple discharge tubes are employed in the final reactor design). The optical flux reaches the wafer through a water-cooled quartz window. The wafer temperature can be controlled in a range from room temperature to 1150°C for seconds up to many minutes.

The gas distribution network for the RTMRPM reactor consists of three gas manifolds and any permissible combinations of a variety of gases (Ar, Ne, N₂, O₂, NH₃, NF₃, forming gas, N₂O, HCl, SF₆, WF₆, heated WCl₆ solid source, H₂, SiH₄, GeH₄, and SiF₄) can be injected into the chamber either through a quartz tube at the bottom of the chamber or through the side port nonplasma injectors. Remote plasma can be generated inside the tube by a microwave discharge cavity operating at 2450 MHz (S band). In contrast to the
conventional localized plasma techniques, the remote microwave plasma approach allows selective and controlled generation of specific plasma species simultaneous with injection of additional nonplasma gases into the process chamber without having to deal with the complications arising from the gas discharge in a composite gas ambient. The availability of remote plasma processing not only allows low-temperature dielectric growth and LPCVD of insulators and silicon epitaxy but also has enabled us to develop several new processes for nonselective deposition of tungsten and its compounds (e.g. nitrides) on insulating layers for MOS gate applications. This system configuration is very flexible for in-situ multiprocessing because it allows rapid cycling of ambient gases, temperature, and plasma with negligible cross-contamination and process memory effects. As a result, this reactor is expected to lead in a new direction towards flexible computer-aided manufacturing (CAM) of custom VLSI circuits.

**Tungsten Deposition Processes**

The main objective of our initial efforts was to develop reliable processes for in-situ fabrication of tungsten-gate MOS devices which requires the growth of gate dielectric by RTO and RTN cycles followed by a nonselective tungsten deposition process to form the gate electrode. Tungsten is quite attractive as an MOS gate material [7]; however, a reliable process for in-situ formation of tungsten gate electrodes has not been developed. Recently, blanket tungsten films were deposited on SiO₂ films at substrate temperatures below 450°C by photo-enhanced and microwave plasma-enhanced (hydrogen plasma) CVD techniques [8]. Tungsten-gate MOS VLSI can be realized if some of the major problems related to the poor adhesion of tungsten to insulating layers, channeling of implanted dopants through tungsten gate, lack of oxidation resistance, and gate dielectric degradation [9] are overcome. We have used two approaches to solve this problem: one by using a silicon glue layer [10] and the other by the use of RTMRPM technology.
As a result of the process limitations of the silicon glue technique [10], a variety of selective and nonselective processes were investigated in this work. Table 1 presents a summary of the tungsten deposition processes developed using our multiprocessing reactor. These techniques are grouped based on the plasma condition and the injection mode of various ambient gases. The depositions were studied extensively in a wide range of gas flows, pressure, and substrate temperature. A number of these nonselective processes are reported for the first time in this work. When WF₆ or a mixture of WF₆+H₂ was injected through the nonplasma port, generation of H₂ plasma, Ar plasma, or Ar+H₂ plasma in the quartz tube promoted nonselective tungsten deposition on insulating surfaces. Addition of Ar to H₂ enhances the plasma emission intensity and density of available atomic hydrogen. Another nonselective deposition technique developed in this work employed WF₆+Ar plasma along with nonplasma H₂. Under appropriate experimental conditions none of these nonselective deposition techniques caused tungsten deposition on the chamber walls or inside the quartz tube. The mixture of NH₃+H₂ and WF₆ always resulted in nonselective deposition for both plasma and nonplasma types of processes. Moreover, the combination of N₂+H₂ plasma and WF₆ also resulted in nonselective metallic film deposition. The films deposited by any of the last three techniques in Table 1 (rows J,K,L) had higher resistivities compared to pure tungsten and were expected to be tungsten nitride compounds. The surface morphology and stability of the CVD tungsten nitrides were functions of the deposition technique and experimental conditions. Tungsten nitride may exhibit useful properties such as oxidation resistance, diffusion barrier, and ion implant channeling stop. Tungsten nitride films could also be formed by RTN of tungsten layers. The films nitrided at the highest temperature (1000°C or more) were powdery; however, the tungsten films nitrided at lower temperatures (e.g. 825°C) were stable. According to the Auger depth profiles the films nitrided at 825°C and above were tungsten oxynitrides. Good in-situ adhesion to insulators was obtained for nonselectively deposited tungsten films thicker than 1 μm.
In some instances, one initial tungsten deposition cycle was followed by another type of deposition in order to obtain optimal adhesion and uniformity properties.

Any combination of WF$_6$/H$_2$/Ar without plasma discharge (rows A through E in Table 1) resulted in very selective tungsten depositions on exposed silicon areas. The selective depositions were performed in a wide range of gas flow rates, pressure, and temperature and selectivity was maintained for depositions well over 1 μm. Compared to a furnace, this single-wafer cold-wall reactor offers a much larger processing window for selective processes without loss of selectivity after long times at elevated temperatures as much as 450°C. SiF$_4$ is known to retard the silicon reduction reaction of WF$_6$. Selectivity was also preserved in a mixture of WF$_6$/H$_2$/SiF$_4$ even at temperatures as high as 650°C. This indicated that in contrast to SiH$_4$, SiF$_4$ cannot initiate nonselective tungsten deposition even at very high deposition temperatures. The mixture of H$_2$ and SiF$_4$ did not result in any silicon deposition at temperatures as high as 650°C.

All of the nonselective deposition techniques developed in this work are applicable to \textit{in-situ} fabrication of metal gate MOS devices. Various MOS devices were successfully fabricated using these techniques. As an example, Fig. 2 plots the high- and low-frequency capacitance-voltage (C-V) characteristics of MOS devices with RTMRPM-deposited tungsten gates and without any final forming gas anneal. Nearly 300 Å thick gate oxide was grown in a furnace in dry oxygen ambient at 950°C for 60 min followed by 60 min Ar anneal at the same temperature. In these particular devices, the initial tungsten nucleation on gate oxide was promoted by a plasma deposition process (row G in Table 1) and this cycle was followed by a nonplasma deposition cycle (row E in Table 1) to make the film thicker. As indicated by the C-V characteristics, the devices exhibit respectable performance and negligible plasma damage. The oxide thickness measured by ellipsometry (308 Å) and extracted from the C-V data (303 Å) were similar which implies that no reduction of SiO$_2$ has occurred during the initial plasma deposition cycle. Assuming negligible fixed oxide
charge density, the flatband voltage value (0.54 V) indicates that the gate work function is located near the silicon midgap which is what should be expected from tungsten and should be ideal for NMOS as well as PMOS devices. The surface-state density distribution plotted in Fig. 3 shows a midgap value of $6 \times 10^{10} \text{eV}^{-1} \text{cm}^{-2}$. This value is relatively low for an unannealed MOS device. There may have been a possibility of some hydrogen annealing during the tungsten deposition process.

**Summary**

In conclusion, this novel rapid thermal/remote microwave plasma multiprocessing technique has potential merits for in-situ fabrication of future high-performance MOS VLSI circuits.

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References


Table 1: Various types of plasma and nonplasma tungsten LPCVD processes developed in the novel RTMRPM reactor.

<table>
<thead>
<tr>
<th>Deposition Type</th>
<th>Gases through the Quartz Tube</th>
<th>Gases through the Side Ports</th>
<th>Microwave Power</th>
<th>Deposition Condition</th>
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</thead>
<tbody>
<tr>
<td>A</td>
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<td>Selective</td>
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<tr>
<td>B</td>
<td>None</td>
<td>WF$_6$+H$_2$</td>
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<td>Selective</td>
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<tr>
<td>C</td>
<td>Ar</td>
<td>WF$_6$+H$_2$</td>
<td>OFF</td>
<td>Selective</td>
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<tr>
<td>D</td>
<td>SiF$_4$</td>
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<td>OFF</td>
<td>Selective</td>
</tr>
<tr>
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<td>H$_2$</td>
<td>WF$_6$</td>
<td>OFF</td>
<td>Selective</td>
</tr>
<tr>
<td>F</td>
<td>H$_2$</td>
<td>WF$_6$</td>
<td>ON</td>
<td>Nonselective</td>
</tr>
<tr>
<td>G</td>
<td>Ar+H$_2$</td>
<td>WF$_6$</td>
<td>ON</td>
<td>Nonselective</td>
</tr>
<tr>
<td>H</td>
<td>Ar</td>
<td>WF$_6$+H$_2$</td>
<td>ON</td>
<td>Nonselective</td>
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<td>WF$_6$+Ar</td>
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<td>Nonselective</td>
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<tr>
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**Figure Captions**

**Figure 1.** Schematic of the novel cold-wall single-wafer lamp-heated rapid thermal/microwave remote plasma multiprocessing (RTMRPM) reactor.

**Figure 2.** High-frequency (100 KHz) and Low-frequency (50 mV/sec) capacitance-voltage characteristics of MOS devices with furnace-grown gate oxide and tungsten gate electrode deposited by RTMRPM. The average flatband voltage across the wafer is 0.537 V (0.096 V standard deviation).

**Figure 3.** Surface-state density distribution of tungsten-gate MOS devices with the C-V characteristics shown in Fig. 2. The average midgap $D_h$ across the wafer is $6.03 \times 10^{10}$ eV$^{-1}$cm$^{-1}$ (0.61$\times 10^{10}$ eV$^{-1}$cm$^{-2}$ standard deviation).
Figure 2

Voltage ramp from -5 V to +5 V

\[ C_{\text{inv}} = 6.6 \text{ pF} \]
\[ C_{\text{dep}} = 2.9 \text{ pF} \]

100 kHz

\[ C_{\max (hf)} = 46.1 \text{ pF} \]
\[ C_{\max (qs)} = 46.2 \text{ pF} \]

Gate Area = 4 \times 10^{-4} \text{ cm}^2

\[ V_{\text{fb}} = 0.520 \text{ V} \]
\[ V_{\text{th}} = -0.346 \text{ V} \]
Surface Potential (V)

$D_{it} (mg) = 4.9 \times 10^{10} \text{ eV} \cdot \text{cm}^{-2}$

50 mV/s voltage ramp from -5 V to 5 V

Figure 3
END
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