DYNAMIC IMPACT RESPONSE BEHAVIOR OF POLYMERIC MATERIALS

FINAL REPORT
(1 May 1976 to 30 April 1977)

August 1977

Prepared Under Contract N00019-73-C-0330

For
Naval Air Systems Command
Department of the Navy
Washington, D.C. 20361

By
Corporate Research and Development
General Electric Company
Schenectady, N.Y. 12301

Approved for public release; distribution unlimited.
The work reported was devoted to making reliable, quantitative impact pulse measurements for free-flight encounters between a spherical projectile and the desired target material. This effort builds on, but also contrasts with, previous studies which involved indentations at programmed incursion rates. Because of the change of techniques involved, developing and verifying the associated instrumented and computation systems were a major part of the effort. The materials of interest continue to be polycarbonate PC, polyc-
methylmethacrylate PMMA, and epoxy resins. Inorganic glass was also used as a target material because impacts can be confidently modeled from theory in this case.

Impact pulse determinations were made on the four target materials with hardened steel strikers of 4.5 mm diameter tips weighing 42 g and of 19 mm diameter weighing 28.3 g usually at 2.5 m per second. The impact strain pulse shapes are increasingly narrower and higher peaked in the order PC, epoxy, PMMA, glass. Semi-quantitative analysis of the general shapes indicates little irreversible deformation work is performed during impacts involving 19 mm striker. However, using the 4.5 mm striker causes irreversible absorption of the net energy received from the striker. The fractional amount absorbed in this series of measurements decreased in the order PC, epoxy, PMMA, and glass.

Computer procedures were developed for analyzing and correcting the detailed pulse shapes to reveal how the force and penetration are related and to follow the evolution of the various energy and momentum contributions. The computer procedures have been verified by using as input model impact pulses, calculated from Hertz elastic theory. However, when the experimentally observed pulse shapes are used as input, the computed force versus penetration behavior is at variance with expectation. Distortion of pulse shape as it travels down the bar is suggested as being responsible. Candidate procedures for dealing with this problem are outlined, additional possible factors are offered, and the implications for future work at greater impact velocities are discussed.
# TABLE OF CONTENTS

## SUMMARY

1. INTRODUCTION
   1.1 Prior Work
   1.2 Present Objectives

2. MATERIALS AND SPECIMEN PREPARATION
   2.1 Polymethylmethacrylate
   2.2 Polycarbonate
   2.3 Epoxy
   2.4 Inorganic Glass

3. IMPACT MEASUREMENTS
   3.1 Impact Apparatus
     3.1.1 The Impact Bar
     3.1.2 Striker
     3.1.3 Electronic Instrumentation
   3.2 Experimental Procedure
   3.3 Auxiliary Measurements
   3.4 Experimental Impact Results

4. ANALYSIS OF DATA
   4.1 Basic Equations
   4.2 Integration Procedures
   4.3 Momentum Measurements
   4.4 Correction for Post-Pulse Behavior
   4.5 Studies Using Inorganic Glass Targets
     4.5.1 Theoretical Elastic Impact Pulse
     4.5.2 Critical Experiments
     4.5.3 Preliminary Fourier Analysis
   4.6 Treatment of Pulse Data
     4.6.1 Detailed Kinematic Analysis
     4.6.2 Macro Kinematic Analysis

5. DISCUSSION

6. ACKNOWLEDGMENT

7. REFERENCES

FIGURES following 7-1

Page

v

1-1

1-2

1-6

2-1

2-1

2-1

2-1

3-1

3-1

3-3

3-5

3-6

3-6

3-7

4-1

4-1

4-4

4-5

4-7

4-9

4-10

4-11

4-14

4-16

4-16

4-18

5-1

6-1

7-1
SUMMARY

This report covers the work done during the twelve-month period beginning May 1976 and is devoted to making reliable, quantitative impact pulse measurements for free-flight encounters between a spherical projectile and the desired target material. This work builds on, but also contrasts with, previous studies which involved indentations at programmed incursion rates. Because of the change of techniques involved, developing and verifying the associated instrumented and computation systems were a major part of the effort. The materials of interest continue to be polycarbonate PC, polymethylmethacrylate PMMA, and epoxy resins. Inorganic glass was also used as a target material because impacts can be confidently modeled from theory in this case.

The experimental procedure involves mounting the specimen on a long PMMA impact bar, which is free to swing in a pendulum-like manner. A striker of desired mass and geometry is similarly arranged to allow an impact with the specimen at a known impact velocity at a controllable spot on the specimen. Momentum change measurements are made, and the strain pulse as it travels down the bar is sensed by a strain gage suitably amplified and recorded. The system was calibrated so as to relate the output signal to the force on the bar.

Impact pulse determinations were made on the four target materials with hardened steel strikers of 4.5 mm diameter tips weighing 42 g and of 19 mm diameter weighing 28.3 g usually at 2.5 m per second. The general pulse shapes are increasingly narrower and higher peaked in the order PC, epoxy, PMMA, glass. A semi-quantitative analysis of the general shape parameters suggests there is little irreversible energy of deformation occurring during any of the impacts involving the 19 mm striker,
but that 25% of the net energy imparted to PC using the 4.5 mm striker was irreversibly absorbed. The fractional amount absorbed in this series of measurements decreased in the order PC, epoxy, PMMA, and glass. The value for the latter is postulated as being zero.

Computer procedures were developed for analyzing and correcting the detailed pulse shapes to reveal how the force and penetration are related and to follow the evolution of the various energy and momentum contributions. These analyses are based on the somewhat simplified assumption that the pulse is uniform over the cross section of the bar. The computer procedures have been verified by using as input model impact pulses calculated from Hertz elastic theory. However, when the experimentally observed pulse shapes are used as input, the computed force versus penetration behavior is at variance with expectation. Distortion of pulse shape as it travels down the bar is suggested as being responsible. Candidate procedures for dealing with this problem are outlined, additional possible factors are offered, and the implications for future work at greater impact velocities are discussed.
DYNAMIC IMPACT RESPONSE BEHAVIOR OF POLYMERIC MATERIALS

1. INTRODUCTION

The response of polymers to impact from small, hard bodies has a wide range of technical and practical implications. As stated in our previous reports, such a seemingly simple event in fact involves a number of scientifically distinctly different but coupled phenomena. The sorting out and determination of the separate phenomena has been a non-trivial exercise. The understanding and quantification of the various processes should provide a basis for (1) achieving better impact resistance and impact resistant materials on a rational basis, (2) diagnosing impact failures and effects, and (3) characterizing and ranking materials.

The present report covers the work done during the twelve-month period beginning May 1976 and continues to focus on polycarbonate PC, polymethylmethacrylate PMMA, and epoxy resins. The work was devoted predominantly to developing and verifying the instrumental and computational systems and to making measurements on specimens subjected to a free-flight encounter by a spherical striker. The materials of investigation were attached to the ends of an instrumented Hopkinson bar. For this purpose, measurements and calculations of the impact process using inorganic glass were undertaken. This has proven to be valuable in terms of providing insight and a rigorous test of the entire experimental/analytical approach. This year's work contrasts from that performed in prior years. Previously the penetrating body was fixed to the movable member of a mechanical test machine. That arrangement allowed the penetration velocity to be held constant or programmed in an independently controlled manner. The present work departs
from the past, both in respect to the instrumentation and the kinds of experiments performed. However, impact velocities in this study were selected to overlap with those used with the mechanical test machines to allow a direct comparison between the two methods.

The present work rests heavily on the results of the four previous yearly programs \((1-4)\). Therefore, this background, especially as it provides perspective to the present effort, will be summarized. The remainder of this report will describe the impact apparatus, present the impact data obtained to date, outline the analytical computer procedures developed to translate the strain pulse impact data into reaction force-penetration information, and finally discuss the results and required future effort. Some portions of the test, e.g., background, description of materials, even though previously reported, but which are needed for completeness, will be incorporated into the present report.

1.1 Prior Work

The sustaining focus of the work has been to understand in depth the processes that occur at the collision site between the polymer of interest and the colliding object. Because there is a large body of information available on the physico-mechanical behavior of PC and PMMA, the approach was to make as much use of this scientific background as possible. Therefore, measurements were initially made at small, constant velocities using mechanical test machines. This work showed that the materials did not behave in the expected classical viscoelastic manner. This rendered some of the analytical structure that we developed inappropriate for treating the experimental results. The observed response of the polymers, even at the smallest detectable penetration depths, exhibited an unexpected form of relaxation behavior. This was coupled with an anelastic yield response mode of significant magnitude.
The yield phenomena, which are revealed from examination of the force-penetration mechanical information, appear to be associated with the development of a lens-like zone of visibly altered material under the contact site. The growth of this "disturbed" zone under the indentation was followed in PC using motion pictures. The zone was found to form virtually from the first moment of contact. The evolution of the zone geometry was quantitatively determined from the photographs. The volume of the zone is about 6±1 times the volume displaced by the indenter. By assuming that the material transformation is describable as an ideal plastic yield process, the (anelastic) yield stress could be estimated using this plus other observations. Two qualitatively different estimates indicate that the yield stress in PC is between 9 and 29 ksi. The term anelastic is used because the disturbed zone and the permanent deformation anneal out completely when the polymer is heated to its glass transition temperature. These somewhat pioneering aspects of the work are expected to be relevant to other processes, such as the forming of polymer shapes, cold flow processing, and concentrated compressive loading of structural polymers.

Because classical viscoelastic theory was found to be inadequate, it was necessary to develop an extensive data base, and measurements of force versus penetration depths have been obtained at velocities ranging from 0.0002 to 6000 inches per minute. A 4.5 mm diameter ball bearing (0.177 inches) has served as the standard size indenter or projectile. However, balls two and four times as large have also been used, as well as cones of various apex angles. The data for PC and PMMA have been found to be representable by a phenomenological "equation of state" for the force F having the form

\[ F = Av^n x^m \]

where v and x are indentation velocity and depth, and A, n, and m are experimentally determined constants.
Considerable attention has been given to understanding the force penetration behavior. For fixed velocity laws of the form $F = Cx^m R^\alpha$ have been proposed to describe the force $F$ behavior in terms of the diameter of the indenter $R$ and the penetration depth $x$. The other terms are characteristic constants. When the material is ideally elastic, $C = (4/3)E/(1-\gamma^2)$, where $E$ is the Young's modulus and $\gamma$ the Poisson ratio of the material; the values of $\alpha$ and $m$ are $1/2$ and $3/2$ in this case, respectively. This is the behavior we expected in polymers at large indentation velocities and/or at small indentation depths. The response at high velocities up to or exceeding 300 m per second is a long-range experimental program goal. The work has focused on getting information in the regimes at which elastic behavior is expected as a reference against which actual behavior can be compared. However, no unambiguously elastic regimes were found. Furthermore, on dimensional grounds $(\alpha + m)$ should equal two. However, this value was generally not observed, implying that additional factors are involved in the impact process. It has become increasingly evident that time variations in velocity have a large effect on the response behavior.

At first glance this conclusion appears to be at variance with the observations made under conditions of constant velocity in which the force required to drive an indenter into the polymer was found to be relatively insensitive to the rate of indentation over a span of 7 1/2 decades of the velocity. However, the reason was not that the apparent elastic modulus was so insensitive to rate, but was due rather to the surprisingly strong dependence of the characteristic stress relaxation time on the prior indentation velocity. At high velocities, the relaxation time was short, whereas at low velocities the relaxation time was long. As a result of these compensating factors for a given penetration depth, the force developed while the indenter is moving is only weakly dependent on velocity.

The pronounced effect of time dependence is revealed most vividly under conditions in which the rate of incursion is abruptly changed or reduced to zero. Such work was done in
detail on PC and PMMA as a part of Contract N00019-75-C-0320 under a broad range of conditions. Pronounced hardening of the material was noted when the material was allowed to rest after having been deformed. That is, the compliance was markedly reduced on restarting the incursion process, after having stopped it temporarily, relative to what it would be if it had not been stopped. These measurements were made as a function of halt duration to determine the effect of history on the material response. Similar measurements were made over a range of depths to define how the relaxation process varies with deformation. Complementary studies were made of the creep recovery of the deformation crater as a function of time under several sets of conditions. These combined results provide a relatively complete picture of the relaxation responses over a broad range of conditions. Applying this information to PC and PMMA, it was possible to correct the observed curves of the force versus depth for relaxation effects. This results in the establishment of the "instantaneous" response curve for these materials, i.e., the inferred response if the material could be deformed infinitely fast.

The load relaxation upon halting the indentation was found to be given over a wide range of times by the unusual equation

\[ F(t) = F(0) (1 + At)^{-B} \]

where \( F(t) \) is load at time \( t \) after stopping the indenter, \( F(0) \) is the load at the instant of stop, and \( A \) and \( B \) are experimental constants. The constant \( A \) is strongly dependent on the indentation rate, and the second constant \( B \) being much less dependent. For the polymer to respond quasi-elastically, no relaxation must occur during the period of loading. This means that the product \( AB \) must become zero. Over the velocity regime explored to date, there is no clear evidence when this will occur. A relatively simple molecular model was formulated which "predicts" the
identical stress relaxation law which had been empirically found to provide the best fit of the relaxation data. The model also predicts the experimentally found proportionality between the constant $A$ and the penetration rate (prior to stopping the indenter). However, further work is needed to test the validity of the model under other sets of conditions, such as sinusoidal loading, free impact, etc.

Although the use of mechanical test machines is convenient, indentations in which velocity is held constant independent of depth do not correspond to the physical event of a free impact. During a collision between a (rigid) projectile and the polymer target material (free impact), the projectile velocity constantly decreases because the indentation forces decelerate the projectile, i.e., the velocity is an implicitly determined variable. Thus, determining the force-time penetration response when the velocity was varied in an independently controlled manner, was undertaken as a problem of intermediate level of complexity. Data on PC and PMMA were obtained for several incursion amplitudes and periods. The above "equation" of state" was partially successful in describing the response.

The work just described sets the stage for undertaking the next level of complexity, viz., determining, analyzing, and finally predicting the response due to free-impact encounters up to velocities of about 300 m per second. Preliminary work showed that in principle an instrumented ballistic impact bar could be used to make the required observations, but that there would be substantial experimental, instrumental, computational, and conceptual challenges in quantitatively bridging the gap between the mechanical test machine work of the past and the ballistic impact work.

1.2 Present Objectives

The original objectives set forth in the work statement were (1) to measure the impact response from a 4.5 mm diameter steel
projectile over the velocity range of 100 to 1000 feet per second, (2) to measure the response when mass, diameter, or other shape variables are altered, (3) to obtain cinemagraphic records of impact testing for use in qualitative and quantitative interpretation, and (4) to develop further theoretical and analytical models of impact behavior to further define the basic constituent physical processes.

During the pursuit of the work, it became clear that the tasks of developing the instrumentation, detecting, processing, and understanding of the results would be more demanding than originally expected. Accordingly, whereas the objectives given above provided the long-term framework, the work can better be described in terms of the following nearer-term goals:

(1) Set up an instrumented Hopkinson bar impact apparatus and demonstrate its capability for detecting the impact pulse with sufficient resolution and accuracy to permit analysis of the data which can provide continuity with the previous work.

(2) Calibrate and establish the reliability of the impact apparatus.

(3) Obtain stress pulses in which the radius, mass, and velocity of the projectile are varied using PC, PMMA, and epoxy as the target materials.

(4) Provide velocity overlap between the upper end of the mechanical test machine experiments and the lower end of the gas gun capability.

(5) Develop the computer software needed to translate the observed stress pulse into force, penetration, energy, momentum transfer, and other related information.

(6) Develop further and critically analyze theoretical models which can permit calculation of the force produced under conditions of varying velocity.

The details of this work are presented in Sections 3 and 4.
MATERIALS AND SPECIMEN PREPARATION

The PC and PMMA materials were unchanged from those used in the previous study. Specimens were used as cut out of sheet stock, making sure that all burrs were removed. The standard specimen size was 1 x 1 x 0.5 inches. The other target materials are described below.

2.1 Polymethylmethacrylate

Specimens were cut from a single sheet of Type G Plexiglas \(^\text{R}\) PMMA (produced by Rohm and Haas). Continuing with our former practice, the sheet was marked off in squares for cutting and a record kept so that the original location of each specimen in the sheet could be identified.

2.2 Polycarbonate

The PC resin was manufactured by the General Electric Company and is designated as Lexan \(^\text{R}\) resin general purpose glazing sheet, Type 9034-112. A single sheet of 0.5 inch thick material was used, and specimens marked, and records kept as for the case of PMMA.

2.3 Epoxy

The epoxy specimens were of previous stock which was molded out of a mixture of 20% by weight of methylene dianiline with Epon \(^\text{R}\) 828 resin manufactured by Shell Chemical Co. Resin was cut into a 0.5 inch slab, cured, and cut into 1.0 x 1.0 inch samples.

2.4 Inorganic Glass

Two types of samples were used having different compositions. One type was a Pyrex glass cut and ground into 0.5 x 1.0 x 1.0 inch samples. The other type was a soda lime glass cut into 0.75 x 0.75 inch samples from sheets 0.022 inches thick.
3. **IMPACT MEASUREMENTS**

This section describes the development of the impact bar apparatus, the various critical components, and the experimental procedure for making the impact observations. This is followed by a description of required certain auxiliary measurements, and finally by a presentation of the stress pulses obtained on PC, PMMA, epoxy, and glass under several conditions.

3.1 **Impact Apparatus**

The overall apparatus is shown schematically in Fig. A. The detailed descriptions of the various components follow.

3.1.1 **The Impact Bar.** Conventionally the materials to be studied under impact, using the instrumented Hopkinson bar technique, are made into a bar. The end is struck by a projectile, and the stress (or strain) pulse detected at some position along its length by a suitable sensor. The motion of the bar is used to measure the momentum transfer.

However, in our work we wish to examine a number of materials under a variety of conditions. Therefore, our plan has been to mount specimens of the materials of interest onto the end of a given impact bar. The samples are held in place by a viscous acoustic coupling substance, which also facilitates the transfer of the pulse generated by the impact to the bar.

In order to minimize any problems due to acoustic impedance mismatch that could cause the signal to be reflected at the specimen/bar interface, the bar material was chosen to be PMMA. The acoustic impedance is given by \( \sqrt{Ep} \), where \( E \) is the Young's modulus and \( \rho \) is density. Since the densities of polymers are nearly constant and since the moduli vary within a factor of about two for the polymers of present interest, the impedances are relatively well matched. This would not be the case if the bar were metal or glass. A further
advantage of using PMMA for the bar is its relatively small value for E. Hence, a correspondingly large strain is developed for a given magnitude of stress. Since a strain gage is used to detect the pulse, less signal amplification is needed, with benefits for the signal-to-noise ratio.

Preliminary measurements indicated that the expected stress pulse duration would be 500 microseconds in length. Since the sound velocity in PMMA is about 2000 m per second, a bar length of at least 1 m is indicated. A bar of about twice this length was selected.

The bar has a square 0.75 x 0.75 inch (19 mm x 19 mm) cross section, is 72 inches (1.83 m) long, and weighs about 800 g. Two SR-4\textsuperscript{R} strain gages are mounted on opposite sides of the bar, about 30 cm from the struck end so as to compensate for possible bar bending due to the impact. A planar horizontal support is provided overhead. This provides a base for attaching fine strings which support the impact bar and the striker in such a way as to permit pendulum swinging motion to occur in the direction of the long axis of the bar. Side-wise motion is prevented by use of a bifilar, V-shaped arrangement in which the bar and also the striker are attached to the apex, and the ends of the arms are attached to the overhead support. The plane of the V is oriented perpendicularly to the long axis of the bar, preventing transverse motion.

By adjusting the strings, the position of the bar is fixed, except for the degree of freedom of motion along its long axis. The bar is suspended 108 cm beneath the overhead support.

The strain gages are embedded in the bar about 2 mm deep, and the necessary electrical leads follow the string supports so as not to interfere with the free swing of the bar. It was found that orienting the strain gages so that they were on the sides of the bar, and not on the top and bottom, was preferable. The reason was that some gravitational bending of the bar was unavoidable, so that some excitation of bending modes in the
vertical plane always occurred. Even so, some horizontal bending or buckling modes were excited. It was, therefore, found to be beneficial to locate the strain gages at a nodal position for this motion, which was located at one-sixth of the bar length. The strain gage and support arrangement is shown in Fig. A, and the effect of moving the gages from an anti-node to a node position is shown in Figs. B and C.

As a means of measuring the motion of the bar caused by the impact, a rider traveling along a taut wire is moved by a metal finger attached to the bar. A steel ruler is mounted behind the wire for convenience in determining the change in position of the rider. This is the quantity $D_{\text{bar}}$ shown in Fig. 3.1A.

3.1.2 Striker. The projectile (striker) was attached to the end of a single bifilar, V-shaped string support, as described in 3.1.1 for the case of the impact bar. By careful location of the upper attachment eyes and adjustment of the two legs of the "V," the impactor could be made to strike the specimen target reproducibly on the longitudinal axis of symmetry of the bar. Different strikers were used. In the case of 12.7 and 19 mm diameter steel ball bearing strikers, the mass was adequate to produce a substantial impact with an approximately 250 cm per second impact. This was produced by allowing the striker to drop about 30 cm from its starting height, shown as $H_{\text{ball}}$ in Fig. 3.1A. As noted in our previous work and when using the gas gun, 4.5 and 9 mm (2.25 and 4.5 mm radii of curvature) diameter projectiles have been standard in our work. Thus, short (ca. 7.5 cm), threaded, hardened steel bars were made, and the ends ground to these radii of curvature. Five threaded attachable masses of 7 g each could be screwed onto the strikers. In this way momenta corresponding to velocities of about 60, 120, 180, 240, and 300 m per second can be imparted to the striker using the standard pendulum geometry. These bar-shaped strikers were fitted with two
bifilar supports to ensure that the striker remained aligned, with its long axis coinciding with that of the impact bar, as well as possible.

The striker was held in the poised position, ready to be released for impact by means of an electromagnet. This facilitated easy measurement of its initial position. A metric ruler positioned along the path of the striker trajectory, along with a camera, positioned for minimum parallax error was used to measure the recoil, $D_{\text{ball}}$ in Fig. 3.1A, of the striker. A cathetometer was provided to measure the height of the striker in its initial and contacting positions.

3.1.3 Electronic Instrumentation. The instrumentation consisted of (1) a strain gage bridge and power supply, (2) a differential amplifier linear to frequencies in excess of 1 MHz to amplify the strain gage signal prior to introducing it to the storage oscilloscope, (3) a digital storage oscilloscope to detect and store the pulse allowing it to be recorded later at slower speed, and (4) a time-base strip chart recorder. These units are described more fully next.

1. The SR- strain gages are of type FAE-25-12S6EL with a gage factor (i.e., relative change in resistance per unit strain) of 2.06±1%. Each gage had a resistance of 120.0±0.2 ohms. Two strain gages were located in non-adjacent arms of a conventional Wheatstone bridge circuit.

2. The strain gage conditioning unit was a Daytronics package comprised of a 9005 main frame power supply and a 9171 strain gage conditioner. This unit provided a stabilized, filtered 5 volt DC excitation voltage to the bridge.

3. A Tektronix differential amplifier model 5A22N was used to amplify the strain gage signal with as little distortion as possible. The bandwidth for this amplifier ranges from DC to 1 MHz and the input impedance is 1 megohm. A fixed gain of about 250 was maintained with a maximum common mode rejection ratio of 100,000:1.
(4) Data accumulation was performed by means of a Nicolet 1090A Explorer digital oscilloscope. This instrument has the capability of detecting and storing up to 4096 12-bit data entries at intervals as close as 1 μsec. Plug-in differential amplifiers, 93A type D, were used to ensure a broad frequency response from DC to 5 MHz. The input impedance is 1 megohm and the common mode rejection ratio is 2500:1.

(5) Permanent data storage was achieved by recording the information gathered by the preceding instrument onto a high impedance Huston Omniscribe® strip chart recorder. This permitted data to be recorded such that the full scale of the paper, 10 inches, corresponds to the full scale on the oscilloscope, and 1 inch on the time base represents 160 μsec.

3.2 Experimental Procedure

The impact experiments in broad outline involve:

(1) Attaching the specimen to the end of the bar using Dow resin 276-V9 to provide acoustic coupling. Visual inspection is made to ensure good overall contact, centering of the specimen relative to the bar and alignment of the bar and striker.

(2) Deducing the initial impact velocity of the striker from its initial height above the point where it contacts the bar, correcting for damping in the pendulum motion.

(3) Determining the recoil of the striker by a similar procedure.

(4) Deducing the recoil velocity of the impact bar from its maximum travel, correcting for frictional effects in the motion of the rider (see 3.1.1).

(5) Measurement of the various masses, permitting a macroscopic test of conservation of momentum to be made.

(6) Initial adjustment of the balance and voltage applied to the strain gage bridge circuit.

(7) Adjustment of the digital storage oscilloscope to ensure proper triggering and suitable time and signal level scale.
(8) Transfer of information from the oscilloscope, after the impact event is recorded and detected, to a strip chart recorder for a permanent record.

(9) Recording, point by point, in some cases of the initial parts of the impact pulse after contact has just been made by the striker. The oscilloscope permits digitized visual readout of the detected signal at 1 microsecond intervals.

(10) Reading the strip chart information by means of a digitizer to translate it into numerical data for computer processing.

3.3 Auxiliary Measurements

For carrying out the computations to be described in Section 4, it is necessary to know the velocity of sound in the impact bar and to have a calibration of the strain gages.

The sound velocity is conveniently measured by simply noting the time \( t \) between the passage of the maximum in the initial compressive impact pulse and its reflection as a corresponding tensile pulse. The velocity is thus simply \( \frac{2L}{t} \), where \( L \) is the distance from the strain gage to the reflecting end of the impact bar. The observed value was 2095 m per second.

Calibration of the strain gage was accomplished by first determining the (long-term) Young's modulus of the bar. Then by dead-loading the bar compressively in the axial direction, the imposed strain is computed from the modulus, load, and bar cross section. This is then compared with the strain signal detected at the storage oscilloscope.

The long-term modulus was determined by bending the bar transversely using known loads, noting the deflection, and calculating the Young's modulus from the well-known beam bending formulas. The measured value was \( 3.13 \times 10^{10} \) dynes/cm².

The strain gage signal was found to be linear with the applied load within experimental error over the explored strain
range, i.e., up to $20 \times 10^{-5}$ cm/cm. This procedure provided a direct means for relating the strain to the oscilloscope output display. This observed calibration factor was $4.29 \times 10^{-7}$ cm/cm per mV at the oscilloscope for the set of instrument settings held fixed during all measurements.

3.4 Experimental Impact Results

Impact measurements were made on Pyrex glass, PC, PMMA, and epoxy targets using 4.5 and 19 mm diameter strikers weighing 42.0 and 28.3 g, respectively. Systematic measurements at this time were made in at least duplicate and, in some cases, triplicate. In all cases except when impacting glass with the 19 mm steel ball, a nominal impact velocity of 2.5 m per second was selected. A lesser velocity of 1.7 m per second was used with the glass to avoid cracking the specimen target.

Representative impact pulses as recorded directly on the strip chart are presented in Figs. A to H. In addition, the average pulse shape parameters are given in Table A. Note that the momentum is proportional to the pulse area. It can be seen that the momentum transferred to the glass was markedly less in the case of the 4.5 mm impact on glass than when the polymers were struck. The broadest pulse occurred using PC, followed by epoxy, and PMMA as measured by any of the peak shape parameters, i.e., time for the peak maximum to occur, time breadth of peak at half maximum strain, total duration of pulse, or momentum. The maximum strains rank in the reverse order. To show typical reproducibility, an example of replicate results is shown superimposed in Fig. G for PMMA. The curves nearly coincide except for small departures at the leading and trailing portions.

As can be seen in the figures, the trailing portions of the pulses often appear to be somewhat oscillatory. The reason for this is not completely determined at this time. This may be due to excitation of bending modes, if the impact is not exactly on center, or may be due to surface waves resulting from Poisson lateral deformations.
These post-pulse waves will probably become more severe problems as the impact velocity is increased. This can be seen from the pulse shown in Fig. I obtained when a PMMA target was struck at about 100 m per second by a 4.5 mm ball shot from a compressed air gun. No attempt has been made to analyze this complex pulse at this time. Analysis of the pulses in terms of the underlying phenomenological physical processes is discussed in Section 4.
<table>
<thead>
<tr>
<th>MATERIAL</th>
<th>GLASS</th>
<th>PC</th>
<th>PMMA</th>
<th>EPOXY</th>
</tr>
</thead>
<tbody>
<tr>
<td>Striker Diameter*</td>
<td>4.5</td>
<td>19</td>
<td>4.5</td>
<td>19</td>
</tr>
<tr>
<td>mm</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Initial Impact</td>
<td>245</td>
<td>174</td>
<td>251</td>
<td>256</td>
</tr>
<tr>
<td>Velocity, cm/sec</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Maximum Strain</td>
<td>6.9</td>
<td>5.2</td>
<td>3.5</td>
<td>4.4</td>
</tr>
<tr>
<td>x 10^4</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Time to Maximum</td>
<td>82</td>
<td>52</td>
<td>197</td>
<td>108</td>
</tr>
<tr>
<td>µsec</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Pulse Duration µsec</td>
<td>352</td>
<td>283</td>
<td>499</td>
<td>354</td>
</tr>
<tr>
<td>Pulse Time Width at</td>
<td>93</td>
<td>53</td>
<td>243</td>
<td>141</td>
</tr>
<tr>
<td>Half Peak Height µsec</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Momentum, g cm/sec</td>
<td>11303</td>
<td>5344</td>
<td>14276</td>
<td>10409</td>
</tr>
</tbody>
</table>

*Mass of 19 mm striker = 28.3 g
"  " 4.5 mm "  " = 42.0 g
4. ANALYSIS OF DATA

4.1 Basic Equations

The basis for deducing the force indentation response in the impact event from the impact pulse was discussed in Section 6.5 of the previous report (4) and is incorporated here with minor revision for convenience.

Because the impact occurs over a concentrated area, the compressive acoustic pulse radiates initially as a spherical wave. However, as it travels down the bar, the compressive wave becomes increasingly planar. If the initial impact does not occur on the axis of symmetry of the bar, then flexural modes can be excited in the bar. Furthermore, surface and transverse waves can be expected as well. Thus, the signal detected at the strain gage comprises a complex superposition of waves. In spite of this, the data will be treated as if the strain pulse consisted only of the planar compressive component. This means the analysis will be somewhat in error and that calibration to provide the necessary corrections would be desirable.

For purposes of analysis, it is convenient to refer to the position of the struck end of the bar as the origin of the z axis and the time \( t = 0 \) as the instant of first contact between the ball and the bar. The ball has a mass \( m \) and is assumed to be incompressible.

Deceleration of the spherical projectile results from the local opposing material reaction force which in turn is supported elastically in the impact bar. Hence, the instantaneous deceleration force is \( A\sigma(t) \) where \( A \) is the cross section of the bar, and \( \sigma \) the area-averaged longitudinal stress. Therefore,

\[
\begin{align*}
\dot{m}z &= - A\sigma(t) \quad \text{Eq. (a)} \\
&= - AE\varepsilon(t) \quad \text{Eq. (b)}
\end{align*}
\]
This can be integrated directly to give

\[ \dot{z} = - \left( \frac{AE}{m} \right) \int_{0}^{t} \varepsilon \, dt + v_o \quad , \quad \text{Eq. (c)} \]

and

\[ z = v_o t - \frac{AE}{m} \int_{0}^{t} \int_{0}^{t'} \varepsilon \, dt' \, dt \quad , \quad \text{Eq. (d)} \]

where \( E \) is the Young's modulus of the bar, \( \varepsilon \) is strain, and \( v_o \) is the initial velocity of the projectile. However, the collision causes the impact bar to be elastically compressed. Therefore, the end of the bar is displaced in the direction of the impacting projectile a distance \( w \)

\[ w = \int_{0}^{z} \varepsilon \, dz \quad , \quad \text{Eq. (e)} \]

or

\[ = c \int_{0}^{t} \varepsilon \, dt \quad , \quad \text{Eq. (e')} \]

where \( c \) is the velocity of sound. Thus, penetration is given by

\[ x = z - w \quad , \quad \text{Eq. (f)} \]

and the velocity of the projectile relative to the end of the bar is given by

\[ \dot{x} = \dot{z} - c \varepsilon \quad , \quad \text{Eq. (g)} \]

Assuming the projectile to rebound with a velocity \( v_f \), its change in momentum must equal the momentum impacted to the bar. This is the total time integral of the force (neglecting reflections), or

\[ m(v_o - v_f) = AE \int_{0}^{\infty} \varepsilon \, dt \quad , \quad \text{Eq. (h)} \]
in which $v_0$ and $v_f$ are the (algebraic) velocities and can have positive or negative values depending on whether the motion is in the direction of the initial impact or its reverse, respectively.

In general the actual measurement of the pulse consists of a strain gage voltage reading as a function of time. This strain is presumably proportional to the strain averaged over the cross section. However, for reasons already discussed, strain at the gage itself may depart from the average strain. Assuming the average planar strain $\varepsilon$ is simply proportional to the observed strain $\varepsilon_g$ at the gage, i.e., $\varepsilon = \beta \varepsilon_g$, we can write for the correction factor $\beta$, using Eq. (h)

$$\beta = \frac{m(v_0 - v_f)}{(AE \int_0^\infty \varepsilon_g \, dt)}$$

Eq. (i)

Wherever $\varepsilon$ appears in the preceding equations, this should be replaced by $\beta \varepsilon_g$. Thus, in principle, the striker velocity $\dot{x}$, penetration velocity $\dot{x}$, penetration depth, resisting force, and other related quantities can be computed from the impact pulse.

Energy is distributed within the specimen and the bar in several ways. Some of the energy is localized near the site of the impact, e.g., that expended in the non-recoverable (anelastic, plastic, viscous) deformation process. There is also a local elastic energy stored in the impact region that subsequently is transferred back to the striker when it rebounds. Finally, there is both a distributed kinetic and potential energy associated with the pulse as it moves along the impact bar. This later energy can be compared as it develops with time. The elastic potential energy is given by

$$U_1 = \int (\text{strain energy/unit volume}) \cdot d(\text{volume})$$

$$= \frac{1}{2} AEC \int_0^t \varepsilon^2 dt$$

Eq. (j)
From the principle of equipartition of energy, this also equals the kinetic energy of the pulse so that the aggregate distributed energy associated with the pulse is

\[ U = AEC \int_0^t \varepsilon^2 dt \quad \text{Eq. (j')} \]

This energy is trapped in the bar, and excites the bar vibrationally, ultimately ending up as heat.

The kinetic energy of the striker can be determined as a function of time. However, we have found no simple way as yet of computing the amount of recoverable energy as a function of time during the course of the collision. Incidentally, neither was there a way to do this in our prior work using the mechanical test machines. The final irreversible work performed on the specimen, of course, is simply the initial kinetic energy of the striker, less its recoil kinetic energy and the pulse energy \( U \).

4.2 Integration Procedures

As can be seen from the preceding section, determination of the various penetration, velocity, energy, and correction quantities requires evaluation of the integrals \( \int \varepsilon dt, \int \varepsilon^2 dt, \int \varepsilon^2 dt \). Once the strain is available in digitized form, the evaluations are straightforward using standard computer techniques. The uncertainties arise primarily from the impact pulse itself. In principle, the pulse should rise from a base line, representing the zero strain condition and should then return to the same base when the striker, upon rebound, breaks contact with the target. However, this idealized situation is usually not observed. The possible reasons for this will be discussed in Section 5. As discussed in 4.1, the value of the correction term \( \beta \) depends on the value of the integral of strain over the duration of the total pulse. Assuming the shape of the
pulse to be correct in other respects, an incorrect assignment of \( \beta \) will lead to such computational aberrations as the striker continuing to exert a reaction force after it has left the specimen. Hence, a way of dealing with this situation is required. This will be considered again in more detail in 4.4 following the presentation of typical computations on the pulse information in Section 4.3.

In summary, starting with the record of the pulse from the strip chart recorder and the experimental observations, the procedure for analyzing the data can be outlined as follows:

1. Determine the momentum change in the striker and the momentum imparted to the impact bar. Compare. Average the values.
2. Convert the recorded pulse into digital form.
3. Translate the digital information into time and strain units.
4. Compute the various time integrals of strain.
5. Determine the calibration factor \( \beta \).
6. Compute the kinematic information such as penetration depth, force, striker translation, striker velocity, pulse energy, bar momentum, as a function of time using the equations in 4.1.
7. Compute correction for the extended long-term tail on the pulse. (See 4.4.)
8. Repeat steps 4 to 7 using corrected pulse.
10. Reconstruct pulse as it would be at site of impact.

Items 8(Alternative) and 9 are explained in Section 4.5

4.3 Momentum Measurements

Determination of the momentum change in an impact is basic to the determination of the correction factor \( \beta \). Conservation of momentum requires that the change in momentum
of the ball must be exactly equal to the momentum imparted to
the impact bar (or other parts of the system).

The procedure used here was to measure both the momentum
change for the striker and for the bar. This was done by
measuring the striker height before it was released and at
the position of impact with the bar. This was done to ±0.1 mm
using a cathetometer fitted with a telescope. A correction was
made for damping in the swing of the striker for a 10 cm initial
swing amplitude as shown in Fig. A. The damping is logarithmic
and is about 5.6% per cycle. Observations on a single cycle at
the amplitudes used for the impact experiments gave a decrement
of 4.0% for the velocity. Thus, in the quarter cycle of a
swing that occurs during an impact test, the velocity is about
1% less than the value calculated using the simple theory.

The damping of the motion of the impact bar was found to
be negligible. However, there was a noticeable correction due
to frictional effects for the rider that marks the horizontal
movement of the bar. The rider, which weighs 0.42 g, would
just move on the wire when it was tipped about 13° from hori-
zontal. Thus, the frictional force was about 100 dynes. Since
momentum is the integral of force over time, a momentum correc-
tion of 51 g cm/sec was computable from the time for the bar
to swing to its maximum amplitude, 0.52 seconds. The estimated
error in reading the horizontal component of the bar movement
is ±0.2 mm. The velocity V of the bar and, thus, the momentum
are calculable from geometries and the simple equations of
motion using

\[ V = \sqrt{2g(R - \sqrt{R^2 - L^2})}^{1/2} \]

where L is the amplitude of the swing, R is the length of the
bar supporting thread, and g is gravitational acceleration.
Even with these refinements, the momentum as computed from the striker data was 2.2% greater than that computed from the bar movement, the range of the absolute standard deviation being ±1.2%. The reason for this systematic bias is not known. The greatest uncertainty in the measurements results from the error in reading the rider position. This is estimated to result in an error of 100 g cm sec\(^{-1}\), which is between 1% and 2% relative error and is less than the systematic value. In making the estimates for \(\beta\), the average value has been used.

4.4 Correction for Post-Pulse Behavior

Ideally an experimentally observed pulse is preceded by a steady, zero-strain signal from the strain gage. The signal should again return to this zero-strain level following the passage of the pulse, i.e., after the striker has rebounded and there is no reaction force possible. In practice a steady pre-pulse signal is usually observed, but the post-pulse level does not always return to this value.

In the case of impacts with the glass targets, the post-pulse signal often exhibits a kind of low amplitude oscillation with a period that approximates the pulse duration. After some "hunting" the value approaches the original pre-pulse level. This is shown in Fig. A for the case of a 4.5 mm diameter, 42 g striker impacting a glass target at 2.45 m per second. In contrast, when the target is a polymer, the post-pulse signal is quite steady, often slowly diminishing and persists for long times at levels appreciably different from the pre-pulse value. The extended pulse resulting from a similar impact as just described but with PC as the target is given in Fig. B.

At this time it is not certain what is causing the extended "tail" in the case of the polymer impacts. Based on our previous work, it seems reasonable that plastic flow or the somewhat delayed strain relaxation in the region of the impact are involved. Residual internal strains in the bar would seem to be
eliminated from consideration because the strains produced in the bars are relatively low <10^{-3}, and the post-pulse behavior depends on the target, whereas the level of strain produced is about the same in all of the impact experiments. This can be seen in Table 3.4A.

However, for purposes of carrying out the various integrations outlined in 4.2 and for getting as "clean" a base pulse for subsequent analysis, a procedure for correcting the pulse for this tail would be beneficial. A procedure for accomplishing this has been developed based on the assumption that (1) the tail is linearly extrapolatable to short times from its behavior at long times, and (2) that for times less than corresponding to the peak strain, the amplitude of the tail component increases proportionally with strain. In addition, we require that at the moment the striker breaks contact with the target upon rebound, force can no longer be exerted on the target.

Thus, a methodology has been developed which is based upon self consistency with the simple laws of nature which require conservation of momentum on the one hand and transfer of momentum only as long as the impacting bodies are in contact. For example, by requiring that the tail amplitude has a given value at some selected time, a slope is defined by this procedure. This provides a description of the tail, which can then be subtracted from the observed pulse to give a corrected pulse.

However, the use of this procedure will become more important after a satisfactory method has been achieved for correcting the pulse for changes of its shape as it travels down the bar. This is discussed in more detail in Sections 4.5 and 4.6.
4.5 **Studies Using Inorganic Glass Targets**

Because of the uncertainty regarding the process that extends the tail of the impact curve and the absence of a priori information as to what the pulse shape should be when a polymer is impacted, the impact between a steel projectile and an inorganic glass specimen was examined in some detail. The completely elastic behavior of glass makes it possible to calculate the pulse shape from the Hertz analysis. The computed pulse shape could then be treated as if it had been experimentally obtained and introduced into the computer programs for computing velocities, penetrations, etc. This procedure provided confirmation of the validity of our computational apparatus. Then by experimentally impacting a glass specimen, we were able to compare the pulse with the theoretically computed pulse. Furthermore, the studies with glass allowed examination of the question of the importance of acoustic impedance matching between the specimen and the impact bar. These points are covered in more detail in the sections that follow.

As an unexpected by-product, we discovered that the glass when struck by the steel striker produced a nearly dead impact, i.e., with very little rebound of the striker. In contrast, a polymer target when struck results in a substantial rebound. This observation was contrary to our intuition, and its examination led to further insight into the impact process.

We had expected the glass target to result in a greater recoil than in the case of the polymer because no significant inelastic processes would be involved. Instead the striker upon impact behaved almost as if it had struck putty. This was a general phenomenon that occurred whenever the target material mounted onto the end of the PMMA impact bar was a stiff elastic material, e.g., silicon carbide, tungsten carbide, hardened steel. Our initial hypothesis was that the acoustic impedance mismatch was somehow preventing the pulse from propagating into the bar.
However, the answer was found to be in the elastic energy expression, Eq. 4.1j', and the \( cc \) term in Eq. 4.1g. The combination of a very stiff target and a relatively compliant impact bar causes substantial compressional shortening of the bar and a large amount of energy to pass into the impact bar. As the striker reacts against the hard target material, it causes the struck end of the bar to accelerate in the direction of the impact. By the time the striker slows down, the end of the bar is moving away from it fast enough for the penetration depth to be substantially reduced at this point in time. Hence, from the Hertz relationship it is clear that the force available to push the striker back is accordingly also reduced. Thus, most of the kinetic energy initially present in the striker is now associated with the pulse that is traveling down the bar.

4.5.1 Theoretical Elastic Impact Pulse. The usual expression for a dynamic elastic impact by a ball on a planar surface is given by relating the deceleration force \( F \) to the indenter travel \( z \) through the Hertz law

\[
F = k z^{3/2}
\]

Eq. (a)

where

\[
k = \frac{4}{3} \sqrt{\frac{R}{v}} \left( \frac{1 - \nu_1^2}{E_1} + \frac{1 - \nu_2^2}{E_2} \right)^{-1}
\]

\( v \) being the Poisson ratio, \( E \) the Young's modulus for the target and striker materials, and \( R \) the radius of curvature of the striker. If \( m \) is the mass of the ball then

\[
- m \ddot{z} = k z^{3/2}
\]

Eq. (b)

This equation can only be solved approximately. However, maximum incursion depths, time of contact, etc., can be found in various standard works. Note that the equations (a) and (b) are reversible. Thus, an ideal loss-less, elastic collision is predicted by this model.
However, as noted in Eqs. 4.1f and g, it is the incursion depth $x$ and not the travel $z$ that determines the force. That is,

$$\dot{m}z = k(z - c\int_0^t edt)^{3/2}$$  

Eq. (c)

should replace Eq. b. This somewhat awkward equation can be readily integrated by computer. The solution in terms of the force versus time and of the striker velocity versus time is given in Figs. A and B. As can be seen, the inclusion of the bar compliance results in a distinctly skewed impulse curve and predicts that the rebound velocity will be much less than the impact velocity even with ideally elastic materials. Thus, the correction for the motion of the target has a major qualitative effect. This computation was made for the example of a 19 mm steel ball impacting a glass target mounted on the end of a PMMA impact rod.

When the theoretically derived impact curve is introduced into the analytical apparatus for deducing the force penetration law responsible for the observed pulse, the Hertz law behavior is deduced. In other words, assumption of the Hertz law allows synthesis of the pulse shape. Analyzing that pulse shape reveals and confirms that the Hertz law governs the pulse. Hence, the analytical procedures and associated computer programs we have developed are valid.

4.5.2 Critical Experiments. As discussed above glass is an ideal medium for testing the impact bar equipment and for allowing straightforward comparison between theory and observation. The work described in this subsection had the dual goals of (1) establishing the validity of the procedure in which specimens are mounted onto the striking ends of bars of another material, and (2) providing a direct comparison between an observed pulse and a pulse computed from theory using the same input conditions.
In Section 3.1.1 the problem of acoustic coupling between the specimen and the bar was briefly discussed. When the target material mounted on the PMMA impact bar is glass, the ratio of the acoustic impedance of the bar to that of the specimen is 1:5. By contrast, when the target is PC or epoxy, the impedance match is typically 3:4. If the bar itself is the target or if the target and bar are made of the same material, there is no mismatch. However, in the latter case, the coupling resin could conceivably cause signal distortion.

In order to test to what extent the signal is affected by less than perfect coupling, the following experiments were performed:

1. A 25 mm diameter Pyrex rod, 1.3 m long, has SR-4 strain gages applied to it, analogously to the PMMA impact rod. The bar was held with the long axis vertical and a 19 mm diameter hardened steel ball dropped on the instrumented end from a height of about 22 mm. This produces an impact velocity of about 1 m per second. The pulse was detected and recorded using the same instrumentation as used with the PMMA bar. Because of the greater stiffness of the glass, more amplification was needed. The resulting pulse is shown in Fig. A.

2. The same instrumented glass bar as used in (1) above had a Pyrex glass specimen mounted on the end to be struck using the standard acoustic coupling resin. The observed pulse produced by a nominally identical impact is shown in Fig. B.

3. The standard PMMA instrumented impact bar was held vertically, a Pyrex glass specimen attached to the end to be struck as in (2) and subjected to the same kind of impact as before. In this case less amplification gain was required. The resultant pulse is shown in Fig. C.

As can be seen, the pulses in (1) and (2) are identical within the noise uncertainty of the measurement. The pulse
in (3), requiring less gain, has considerably less relative noise content, but otherwise appears to be identical with the other two. Thus, except for possible subtleties obscured by the noise, there are no major qualitative differences attributable to the method of mounting the specimen to the acoustic match of the impact bar with the target specimen. It may be noted that there was a strong rebound of the striker in experiments (1) and (2) but a "dead" response in (3).

The second major reason for measuring the impact response of glass was to investigate the reason for the observed soft response during the very initial stages of impact. This was noted in preliminary experiments with PC and PMMA specimens. As may be recalled from our earlier work using the mechanical test machines, deciding when the first moment of contact occurs is often difficult. Based on the assumed validity of the Hertz law, plotting \((\text{force})^{2/3}\) versus time and extrapolating to zero force often provided an unambiguous determination of the time of first contact. The slope of this curve is related to the compliance of the material. In the case of the present experiments using the polymer targets mounted on the impact bar, a similar procedure could be followed using the initial portions of the pulse. However, the slope was anomalously low by about an order of magnitude. Therefore, the determination of the pulse shape for glass, in which the Hertz constant is unique because of its near-ideal-elastic behavior, provides a powerful tool for investigating this effect. Furthermore, it would be obviously satisfying to get good detailed agreement between the observed pulse with that calculated from theory.

The observed and the theoretical pulses corresponding to a 19 mm steel ball striking a Pyrex specimen attached to a PMMA bar with an impact velocity of 1.74 m per second is given in Fig. D. As can be seen, there is good correspondence in the general shape in the vicinity of the maximum. However, the gradual tapering ramp on the leading edge of the experimental pulse is not predicted theoretically.
Whereas the theoretical pulse is that calculated for the struck end of the bar, actually the pulse is detected about 30 cm from the site of the impact. Hence, some distortion of the pulse can be expected as it travels due to dispersion and attenuation effects. Probably the most direct way to examine this would be to analyze the pulse in terms of its various frequency components. The pulse could then be reconstructed at some other position in time and space after applying the appropriate adjustments for the phase and amplitude values for each component. Some preliminary work in this direction is discussed in the following subsection. It is possible that other factors may contribute significantly to the (presumed) alteration in pulse shape, such as reflections, different propagation modes, etc.

4.5.3 Preliminary Fourier Analysis. As is well known, within very broad limits, any finite function of some independent variable, such as time, definable over some range of that variable, can be expressed as a Fourier sine, cosine, or both series. An alternative representation, employed here, is to express the function \( f(t) \) in the form

\[
f(t) = \sum_{n=0}^{\infty} A(n) \sin (2\pi n \nu_o t + \delta_n) \quad \text{Eq. (a)}
\]

where \( A(n) \) is the amplitude of the \( n \)th harmonic of the fundamental frequency \( \nu_o \), and \( \delta_n \) is the phase shift for that harmonic. Thus, \( A(n) \) represents the strength or relative contribution of a given frequency to the total function. If \( f(t) \) represents the shape of the impact pulse, then the values of \( A(n) \) and \( \delta_n \) can be computed by well-known procedures.

If dispersion occurs, then each component frequency will travel at a somewhat different velocity. This means that if a signal is known at a given position along the impact bar, the signal at some more distant position can be determined by an appropriate adjustment of the phase, viz., by subtracting the quantity \( 2\pi n \nu_o L/C(n \nu_o) \) from \( \delta_n \). The term \( L \) is the
distance between sensing locations, and \( C(nv_o) \) is the sound velocity as a function of component frequency \( nv_o \). By noting that Young's modulus \( E = \rho c^2 \) where \( \rho \) is density, the dependence of the sound velocity \( C \) on frequency can be determined from the 1974 report (3), Fig. 7A. Over the range of current interest, this can be given by

\[
C = 1.737(1 + 0.0481 \ln(v))^{1/2} \text{ m/sec} \quad \text{Eq. (b)}
\]

Preliminary Fourier analysis of a theoretically computed impact pulse has been performed by using a modification of a previously developed computer routine. The pulse was then shifted to a position 30.5 cm down the impact bar (corresponding to the position of the strain gage relative to the specimen force), using the procedure outlined in the previous paragraph, and using Eq. (a) to describe the dispersion. However, it was quickly recognized that there is considerable arbitrariness in this procedure because a Fourier representation predicts that the function repeats with a period corresponding to the fundamental frequency \( v_o \). However, the pulse is a single event. That is, the strain gage senses a long period of no signal followed by the pulse, followed (ideally) by another long period of no signal. Therefore, for purposes of the Fourier analysis, the periods of no signal are part of the function to be analyzed. Therefore, the procedure was to add a null interval both preceding and following the calculated impact pulse. The results are given in Fig. A. The one curve shows the initial pulse calculated for the case of a 4.5 mm projectile having a mass of 42 g and an initial velocity of 150 cm per second, impacting a glass target mounted on a PMMA impact bar. The other two curves show the predicted shapes of the pulse detected 30.5 cm down the bar when the preceding and trailing null signals are 10 and when they are 200 microseconds each.
As can be seen, the predicted pulse shape is a sensitive function of the duration of the null portions relative to the actual pulse duration. Since, in principle, the null portions extend indefinitely, extending the null regions to infinity transforms the Fourier series analysis to a Fourier integral analysis. Another approach is to examine the behavior as the null portions are made increasingly longer. However, the number of computations increases as the square of the total length of the signal to be analyzed. Hence, even using a computer, this approach becomes awkward, and suggests that it may be necessary to use the Fourier integral approach instead.

In any case, dispersion appears to have the expected effect of extending the leading edge of the pulse as it moves down the impact bar. More work is needed to establish whether this accounts for the departures of the experimentally observed pulses from the theoretical computed ones, as shown in Fig. 4.5.2D.

4.6 Treatment of Pulse Data

4.6.1 Detailed Kinematic Analysis. Sections 4.1 and 4.2 discussed the basic equations needed to convert the pulse curves into information regarding reaction force, penetration depths and velocities, ball velocity, various energy and momentum terms, etc. The observed pulse results from an impact between a steel striker and a glass target fixed to a PMMA bar have been analyzed as have the experimental pulse data for two cases when the target was PMMA. In addition, theoretical pulses for the steel glass encounter were computed under identical impact conditions as were experienced in the experiments. The remaining data have not been processed as yet because it appears at this time that some kind of a pulse shape correction, such as discussed in Section 4.5.3, will need to be applied in order
to get correspondence between the pulse shape and a valid kinematic analysis. This will be discussed in more detail later.

Table 3.4A presented the average results for several replicate measurements. However, the analyses require as input the results of particular runs. The designation for a 4.5 mm diameter, 42 g, and for a 19 mm diameter, 28.3 g steel striker impacting glass at 2.45 and 1.74 m per second, respectively, are GL3N-2 and GL1N-1. The resulting pulses have been analyzed and compared with the results derived from the theoretically expected in Table 4.6A. Specimen computer analyses for the case of the GL1N-1 impacts are given in Figs. A and B and the deduced force versus penetration relationships is shown in Fig. C for the same examples.

The analysis of the theoretical pulse reproduces, within certain errors due to numerical integration procedures, the Hertz force law assumed in the first place. However, as can be seen in Fig. C, the force law deduced from the experimentally observed pulse is quite different. As already mentioned, the apparent compliance in the experimental case is initially much greater than expected. This is believed to be caused by a distortion of the pulse due to attenuation and/or dispersion effects. Analysis on the distorted pulse leads to a fictitious force law.

Nevertheless, in many other respects, there is good agreement between the theoretical and experimental results. Striker recoil velocities are close to predicted values; hence the momenta are also in good agreement. The predicted peak force is about 20% greater than the observed value. However, it appears earlier with respect to the first detectable departure from a zero signal. Although the observed pulse width is about the expected value, the observed pulse tail stretches out to much longer times. The theoretical analysis shows that the local stored energy in the target
is completely expended in causing the striker to rebound. In contrast, the analysis of the experimental data indicates that about 10% of the initial kinetic energy of the striker remains in the impact bar after the rebound has occurred, in some form other than being associated with the traveling impact pulse.

These deviations between the experimental results and theoretical expectations, under conditions in which nearly ideal impacts are experimentally expected, was interpreted as indicating that the full analysis of the pulse data is premature at this time. Nevertheless, analysis of representative polymer data has been performed on runs PM3N-2 and PM1N-3 in which a 4.5 mm diameter, 42 g, and 19 mm, 28.3 g steel strikers having initial velocities of 2.48 and 2.56 m per second, respectively, were used to produce the impacts. These results are also given in Table A.

As can be seen the recoil velocity in the case of PMMA is appreciably greater and reflects itself in the greater momentum transfer. The peak force is substantially less and the pulse width correspondingly greater than is the case for the glass impact. This is expected from the greater elastic modulus for glass. It is interesting to note in the case of the polymer that the peak force develops earlier than the maximum penetration. This may simply be a computational artifact, or it may reflect the relaxational processes expected to occur in PMMA. Finally, the residual localized energy in the case of the 4.5 mm striker is nearly twice that calculated for any of the other experimental pulses and probably reflects a real non-reversible energy absorption.

4.6.2 Macro Kinematic Analysis. Because the fine structure of the pulse causes certain complications in the detailed analysis of force, depth, energy, etc., versus time, it is of interest to examine what kind of information can be obtained from less detailed descriptions of the pulse data. Such information was presented in Table 3.4A for the three polymers
TABLE 4.6.1 A
Kinematic Analysis of Pulses for Glass and PMMA

<table>
<thead>
<tr>
<th>TARGET</th>
<th>GLASS</th>
<th></th>
<th>PMMA</th>
<th></th>
</tr>
</thead>
<tbody>
<tr>
<td>Striker Diameter, mm</td>
<td>4.5</td>
<td>19</td>
<td>4.5</td>
<td>19</td>
</tr>
<tr>
<td>Striker Mass, g</td>
<td>42.0</td>
<td>28.3</td>
<td>42.0</td>
<td>28.3</td>
</tr>
<tr>
<td>Impact Velocity, cm/sec</td>
<td>245</td>
<td>174</td>
<td>248</td>
<td>25.6</td>
</tr>
<tr>
<td>Experiment No.</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Recoil Velocity, cm/sec</td>
<td>Theory</td>
<td>Obs.</td>
<td>Theory</td>
<td>Obs.</td>
</tr>
<tr>
<td></td>
<td>GL3N-2</td>
<td></td>
<td>GL1N-1</td>
<td></td>
</tr>
<tr>
<td></td>
<td>23.9</td>
<td>25.4</td>
<td>17.6</td>
<td>15.3</td>
</tr>
<tr>
<td></td>
<td>11294</td>
<td>11357</td>
<td>5422</td>
<td>5357</td>
</tr>
<tr>
<td>Momentum Transfer g cm/sec</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td>Theory</td>
<td>Obs.</td>
<td>Theory</td>
<td>Obs.</td>
</tr>
<tr>
<td></td>
<td>PM3N-2</td>
<td></td>
<td>PM1N-3</td>
<td></td>
</tr>
<tr>
<td></td>
<td>72.0</td>
<td>80.7</td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td>13440</td>
<td>9529</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Peak Force, Kg</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td>142</td>
<td>117</td>
<td>100</td>
<td>85</td>
</tr>
<tr>
<td></td>
<td>43</td>
<td>76</td>
<td>31</td>
<td>51</td>
</tr>
<tr>
<td></td>
<td>43</td>
<td>75</td>
<td>31</td>
<td>51</td>
</tr>
<tr>
<td></td>
<td>78</td>
<td>89</td>
<td>53</td>
<td>54</td>
</tr>
<tr>
<td></td>
<td>183</td>
<td>&gt;363</td>
<td>124</td>
<td>&gt;290</td>
</tr>
<tr>
<td>Time to Peak Force</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td>144</td>
<td>84</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Time to Max. Penetration</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td>147</td>
<td>88</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Pulse Width at Half Peak</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td></td>
<td>182</td>
<td>102</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Pulse Duration</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Residual Non-Pulse Energy</td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>Initial K.E. of Striker</td>
<td>▼10⁻³</td>
<td>.08</td>
<td>▼10⁻³</td>
<td>.11</td>
</tr>
<tr>
<td></td>
<td>.22</td>
<td>.07</td>
<td></td>
<td></td>
</tr>
</tbody>
</table>

Note: Time units are µsec.
investigated and glass. For example, if an approximate measure of the final value of the integrals \( \int \varepsilon \, dt \) and \( \int \varepsilon^2 \, dt \) could be established, then it may be possible to estimate the non-recoverable energy that remains localized in the region of the impact.

For this purpose we note that \( E_1 \), the energy extracted from the striker, i.e., its initial kinetic energy less its rebound energy is given by

\[
E_1 = p(V_o - \frac{P}{2m}) \tag{a}
\]

where \( p \) is the measured change in momentum resulting from the impact, \( V_o \) is the velocity at impact of the striker and \( m \) is its mass. Defining \( Q = t_0.5 \cdot \varepsilon_{\text{max}} \) and \( J = t_0.5 \cdot \varepsilon^2_{\text{max}} \), we also consider the possibility that

\[
k_q Q \approx \int_0^\infty \varepsilon \, dt \tag{b}
\]

and

\[
k_j J \approx \int_0^\infty \varepsilon^2 \, dt \tag{c}
\]

where \( \varepsilon_{\text{max}} \) is the measured peak pulse strain, \( t_0.5 \) is the time breadth of the pulse at half the peak strain, and \( k_q \) and \( k_j \) are proportionality constants to be determined. As given in Eq. 4.1h

\[
p = AE \int_0^\infty \varepsilon \, dt
\]

and introducing this into Eq. (b) gives:

\[
k_q = \frac{p}{AEQ}
\]

Similarly from Eq. 4.1j'

\[
U_f = U(\infty) = AE \int_0^\infty \varepsilon^2 \, dt
\]
which when combined with Eq. (c) gives

\[ k_j = \frac{U_f}{AEcJ} \]

The value of \( U_f \) is not known without resorting to a detailed pulse analysis. However, in the case of glass, it can be assumed that there is no non-recoverable energy left in the glass following the impact. Hence, specifically for glass

\[ U_f = E_1 \quad \text{or} \quad k_j = \frac{E_1}{AEcJ} \]

This cannot be assumed a priori for the other materials. Therefore, the impacts with glass will be assumed to provide a means of calibration.

The experimental results from Section 3.4 are analyzed in terms of the above discussion in the following table. The values for the bar modulus \( E \), the sound velocity \( C \), and the cross-sectional area \( A \) are \( 5.18 \times 10^{10} \) dynes/cm\(^2\), \( 2.095 \times 10^3 \) m/sec, and \( 3.61 \) cm\(^2\), respectively, were used when needed to compute the parameters given in the table.

The results shown in the table for \( k_q \) and \( k_j \) suggest that this macro approach can give interpretable quantitative information, and the constancy of \( k_q \) indicates that the impact pulses more or less scale in a regular way. The major departure occurs in the case of the impact on glass at low velocity. The two impacts on glass give values for \( k_j \) that differ by merely 5% for rather disparate impact conditions. Taking the average value 0.74 allows computation of the ratio of the energy estimated to be tied up with the traveling pulse relative to the energy transferred to the target and impact bar, i.e. \( (k_jAEcJ)/E_1 \). These results suggest that the impacts using the 19 mm striker are substantially reversible, if not elastic. However, with the 4.5 mm striker, the amount of irreversible work increases in the order \( G<PM<E_P<PC \). This seems to be in accord with expectation.
<table>
<thead>
<tr>
<th>Table 4.6.2 A Analysis of Impact Using General Pulse Shape Parameters</th>
</tr>
</thead>
<tbody>
<tr>
<td><strong>Striker Diameter, mm</strong></td>
</tr>
<tr>
<td>---</td>
</tr>
<tr>
<td>4.5</td>
</tr>
<tr>
<td>251</td>
</tr>
<tr>
<td>3.5</td>
</tr>
<tr>
<td>243</td>
</tr>
<tr>
<td>14.3</td>
</tr>
<tr>
<td>851</td>
</tr>
<tr>
<td>29.8</td>
</tr>
<tr>
<td>0.75</td>
</tr>
<tr>
<td>25</td>
</tr>
</tbody>
</table>

**Note:** The table data is extracted from the image and formatted to enhance readability. The values in the table represent specific parameters related to impact analysis using general pulse shape parameters. The table includes columns for various impact parameters such as striker diameter, striker mass, target velocity, maximum stress, time to peak stress, peak pressure, and more, each corresponding to different units and scales.
5. DISCUSSION

The data and analyses presented in Sections 3.4, 4.5, and elsewhere demonstrate that high quality, reproducible impact pulses can be produced and that the computational tools developed for analyzing the pulses work properly. Summarizing briefly, impact measurements have been made on PC, PMMA, epoxy, and glass. The measurements on the polymers show that for a given intensity impact the compliance and amount of irreversible deformation increase in the order glass<PMMA<epoxy<PC. These conclusions are based on the macro-scale analysis of the pulse data and are in accordance with our general expectations. However, the detailed analyses of the force as a function of penetration, and penetration as a function of time are at variance with the results of our prior work and also with the predictions based on a theoretical modeling of purely elastic collisions.

The use of glass as a target material has been particularly useful in defining the nature of the problem related to the detailed analysis. Measurements on this material provide a reference base for establishing the correspondence attainable between the calculated results and actual material behavior. It is almost certain that a basis can be found by which the observed pulse can be corrected for the attenuation and dispersion effects, even though the correction formulae are not now firmly known. However, other factors must also be considered, such as contributions from surface elastic waves, shear waves, and anelastic/plastic effects.

It has already been noted and as shown by comparing Fig. 3.41 with 3.4H, the pulse becomes more complex at higher impact velocities. Increasing the impact velocity increases the content of the higher frequency components. This may require consideration of the dimensions of the impact bar's small dimension relative to the wave length of a given acoustic component. If $\lambda$ is large relative to the minor dimension of the bar, then the use of Young's modulus to
define velocity is valid. If \( \lambda \) is relatively small, then the bar acts as a bulk solid and the bulk modulus is needed. For the case of PMMA, the sound velocity corresponding to transmission in bulk material is some 40% greater than for transmission down a long, thin bar. Thus, bar geometry can affect the dispersion corrections.

These kinds of considerations will need to be addressed in seeking to describe fully the impact process at increasingly higher velocity. The degree of refinement practically attainable will be a function of the ease of its achievement and need, and will be determined largely by experience. However, it is gratifying to note that even without a detailed analysis, much qualitatively useful information can be derived from observations of the general pulse characteristics as was discussed in Section 4.6.2.
6. ACKNOWLEDGMENT

The principal investigator acknowledges the substantial contributions of his collaborators and colleagues. The development and installation of the experimental apparatus and instrumentation and the experimental measurements were largely the accomplishments of G.F. Selden, who also developed various software and computer techniques and then used these and other programs in processing the data. Valuable consultation, guidance and assistance were provided by W.P. Minnear, especially during the troubleshooting phases of the work. Counsel and help with the electronic instrumentation from R.L. Mehan and from D.B. Sorensen and L.H. String are gratefully acknowledged.
7. REFERENCES


Fig. 3.1A. Impact apparatus.
Fig. 3.1.1A. Support arrangement of impact apparatus.
Fig. 3.1.1B. Output signal with two strain gages not located at node position.
Fig. 3.1.1C. Output signal with two strain gages located at nodal positions.
Fig. 3.4A. Strain pulse produced by impact of 19 mm diameter striker with glass target at a velocity of 174 cm/sec.
Fig. 3.4B. Strain pulse produced by impact of 19 mm diameter striker with PC target at a velocity of 256 cm/sec.
Fig. 3.4C. Strain pulse produced by impact of 19 mm diameter striker with PMMA target at a velocity of 258 cm/sec.
Fig. 3.4D. Strain pulse produced by impact of 19 mm diameter striker with epoxy target at a velocity of 260 cm/sec.
Fig. 3.4E. Strain pulse produced by impact of 4.5 mm diameter striker with glass target at a velocity of 245 cm/sec.
Fig. 3.4F. Strain pulse produced by impact of 4.5 mm diameter striker with PC target at a velocity of 251 cm/sec.
Fig. 3.4G. Strain pulse produced by impact of 4.5 mm diameter striker with PMMA target at a velocity of 248 cm/sec. Solid curve is PM3N-1; the dashed curve is PM3N-2.
Fig. 3.4H. Strain pulse produced by impact of 4.5 mm diameter striker with epoxy target at a velocity of 250 cm/sec.
Fig. 3.41. Strain pulse produced by impact of 4.5 mm diameter striker with PMMA target at a velocity of 100 m/sec.
Fig. 4.3A. Damping behavior of striker.
Fig. 4.4A. A typical pulse obtained using a glass target showing long-term post-pulse signal.
Fig. 4.4B. A typical pulse obtained using a PC target showing long-term post-pulse signal.
Fig. 4.5.1A. Force versus time for elastic impact of 19 mm steel ball on glass target mounted on PMMA bar at velocity of 1 m. Solid line is without allowing for bar displacement. Dashed line is correct curve including bar displacement effect.

Fig. 4.5.1B. Striker velocity versus time for impact described in Fig. A above.
28 g Steel Ball / Glass Bar / 100 cm/sec
Fig. 4.5.2B

28 g Steel Ball / Glass Specimen / Glass Bar / 100 cm/sec
Fig. 4.5.2D. Impact pulse due to a 19 mm steel ball striking a Pyrex target on a PMMA bar.
Fig. 4.5.3A. Effect of adding null signals of indicated duration (μsec) on computed shape of a distance-shifted theoretical impact pulse; only dispersion effects are considered.
**Fig. 4.6A.** Computer analysis of experimental pulse for GLIN-1.
<table>
<thead>
<tr>
<th>TIME (SEC)</th>
<th>DEPTH (IN.)</th>
<th>PENETRATION VELOCITY (IN./SEC)</th>
<th>FORCE (LBS)</th>
<th>PULSE ENERGY</th>
<th>LOCAL ENERGY</th>
<th>TOTAL ENERGY</th>
<th>INSTANTANEOUS VELOCITY</th>
</tr>
</thead>
<tbody>
<tr>
<td>0.000</td>
<td>0.640</td>
<td>0.172</td>
<td>0.05</td>
<td>0.015</td>
<td>0.015</td>
<td>0.015</td>
<td>0.15</td>
</tr>
<tr>
<td>0.100</td>
<td>0.640</td>
<td>0.172</td>
<td>0.05</td>
<td>0.015</td>
<td>0.015</td>
<td>0.015</td>
<td>0.15</td>
</tr>
<tr>
<td>0.200</td>
<td>0.640</td>
<td>0.172</td>
<td>0.05</td>
<td>0.015</td>
<td>0.015</td>
<td>0.015</td>
<td>0.15</td>
</tr>
<tr>
<td>0.300</td>
<td>0.640</td>
<td>0.172</td>
<td>0.05</td>
<td>0.015</td>
<td>0.015</td>
<td>0.015</td>
<td>0.15</td>
</tr>
<tr>
<td>0.400</td>
<td>0.640</td>
<td>0.172</td>
<td>0.05</td>
<td>0.015</td>
<td>0.015</td>
<td>0.015</td>
<td>0.15</td>
</tr>
<tr>
<td>0.500</td>
<td>0.640</td>
<td>0.172</td>
<td>0.05</td>
<td>0.015</td>
<td>0.015</td>
<td>0.015</td>
<td>0.15</td>
</tr>
<tr>
<td>0.600</td>
<td>0.640</td>
<td>0.172</td>
<td>0.05</td>
<td>0.015</td>
<td>0.015</td>
<td>0.015</td>
<td>0.15</td>
</tr>
<tr>
<td>0.700</td>
<td>0.640</td>
<td>0.172</td>
<td>0.05</td>
<td>0.015</td>
<td>0.015</td>
<td>0.015</td>
<td>0.15</td>
</tr>
<tr>
<td>0.800</td>
<td>0.640</td>
<td>0.172</td>
<td>0.05</td>
<td>0.015</td>
<td>0.015</td>
<td>0.015</td>
<td>0.15</td>
</tr>
<tr>
<td>0.900</td>
<td>0.640</td>
<td>0.172</td>
<td>0.05</td>
<td>0.015</td>
<td>0.015</td>
<td>0.015</td>
<td>0.15</td>
</tr>
</tbody>
</table>

Fig. 4.6B. Computer analysis of theoretical pulse for GLIN-1.
Fig. 4.6C. Calculated force versus penetration depth computed from impact pulse using glass target.

Experiment 341N-1.
DISTRIBUTION LIST

Contract No. N00019-76-C-0330

Naval Air Systems Command
Attn: Code AIR-52032
Washington, DC 20361

Naval Air Systems Command
Attn: Code AIR-604
Washington, DC 20361

Office of Naval Research
Washington, DC 20361
Attn: Code 471
Code 472
Code 473

Naval Ordnance Laboratory
White Oak, MD 20910
Attn: Code 2301
Code 234

Materials Sciences & Engineering Laboratory
Stanford Research Institute
Menlo Park, CA 94025

Institute for Materials Research
National Bureau of Standards
Washington, DC 20234

Battelle Memorial Institute
505 King Avenue
Columbus, OH 43201

IIT Research Institute
Attn: Ceramics Division
10 West 35th Street
Chicago, IL 60616

Naval Ship R&D Center
Washington, DC 20007

Naval Ship R&D Center
Annapolis Lab. (Code 287)
Annapolis, MD 21402

Air Force Materials Laboratory
Wright-Patterson Air Force Base
Dayton, OH 45433
Attn: LN 1 LAE 1

Office of Naval Research
Washington, DC 20361
Attn: Code 471

Naval Ordnance Laboratory
White Oak, MD 20910
Attn: Code 2301
Code 234

Materials Sciences & Engineering Laboratory
Stanford Research Institute
Menlo Park, CA 94025

Naval Ordnance Laboratory
White Oak, MD 20910
Attn: Code 2301
Code 234

Atomic Energy Commission
Technical Information Service
P.O. Box 62
Oak Ridge, TN 37830

NASA Headquarters
Attn: B.G. Achhammer
Washington, DC 20546

NASA-Lewis Research Center
Attn: R.F. Lark, Mail Stop 49-1
21000 Brookpark Road
Cleveland, OH 44135

Army Materials & Mechanics Research Center
Watertown, MA 02172
Attn: Dr. R.N. Katz
Dr. G. Thomas

U.S. Army Aviation Material Laboratories
Fort Eustis, VA 23604

Effects Technology, Inc.
Attn: F.R. Tuler
P.O. Box 30400
Santa Barbara, CA 93105
Naval Undersea R&D Center
San Diego, CA 92117

Engineering Experiment Station
Georgia Institute of Technology
Atlanta, GA 30332

Goodyear Aerospace Corporation
Attn: G. Wintemute
Litchfield Park, AZ 85340

Pratt & Whitney Aircraft Div.
United Aircraft Corporation
Attn: C. C. Goodrich
East Hartford, CT 06108

TRW Equipment Lab
23555 Euclid Avenue
Cleveland, OH 44117

Aero-Electronic Technology Dept.
Attn: George Tatnall
Naval Air Development Center
Warminster, PA 18974

Vought Aeronautics Division
Attn: A. E. Hohman, Jr.
LTV Aerospace Corporation
P.O. Box 5907
Dallas, TX 75222

Engineered Fabrics Division
Goodyear Aerospace Corporation
Akron, OH 44315

Bell Aerosystems Company
Attn: N. E. Wahl
Buffalo, NY 14240

Hycronautics, Incorporated
Pindell School Road
Howard County
Laurel, MD 20810

Brunswick Corporation
Marion, VA 24354

1 Plastics Technical Evaluation Center
Picatinny Arsenal
1 Dover, NY 07801

University of Dayton
Library on Materials Research
300 College Park Avenue
Dayton, OH 45409

University of Michigan
1 Attn: F. G. Hammitt
Dept. of Nuclear Engineering
Ann Arbor, MI 48104

Olin Corporation
1 Chemical Group
New Haven, CT 06540

Aerospace Corporation
1 Materials Laboratory
P.O. Box 95085
Los Angeles, CA 90045

Air Force Avionics Laboratory
1 Wright Patterson AFB
Dayton, OH 45433
Attn: AVTL

U.S. Army Research Office
1 Box CM, Duke Station
Durham, NC 27706

Applied Technology Division
Avco Corporation
1 Lowell Industrial Park
Lowell, MA 01851

Solar Division
1 Attn: Dr. A. G. Metcalfe
International Harvester Company
2200 Pacific Highway
San Diego, CA 92112

Material Sciences Corporation
1 17777 Walton Road
Blue Bell, PA 19422
University of Illinois
Attn: Prof. H. T. Corten
Dept. of Theoretical & Applied Mech.
Urbana, IL 61801

Naval Ship Engineering Center
Attn: Code 6101E
Washington, DC 20361

Westinghouse Research Labs.
Attn: R. Bratton
Beulah Road
Pittsburgh, PA 15235

E. I. du Pont de Nemours & Company
Attn: C. Zweben, Bldg. 262
Textile Fibers Dept.
Carothers Research Lab
Experimental Station
Wilmington, DE 19888

NTDSC
Southwest Research Institute
P.O. Drawer 28510
San Antonio, TX 78284

Texas A&M University
Attn: Prof. J. L. Rand
Aerospace Engineering Dept.
College Station, TX 77843

Dr. George C. Chang
Advanced Physical Methods Branch
Division of Energy Storage Systems
Office of Conservation
U.S. Energy Research and Development Administrations
Washington, DC 20545

Prof. D. Uhlmann
Massachusetts Institute of Technology
Cambridge, MA 02139

1 Dr. S. S. Sternstein
Rensselaer Polytechnic Institute
110 8th Street
Troy, NY 12181

Prof. R. Doremus
1 Rensselaer Polytechnic Institute
110 8th Street
Troy, NY 12181

1 Prof. M. Goldstein
Belfer Graduate School
Yeshiva University
500 W. 185 Street
New York, NY 10033

John D. Ferry
Department of Chemistry
University of Wisconsin
Madison, WI 53706

Library
1 National Bureau of Standards
Washington, DC 20234

Prof. J. H. Gibbs
Metcalf Chem. Lab.
1 Brown University
Providence, RI 02912

Polymer Research Institute
Univ. of Massachusetts
1 Amherst, MA 01002

Dr. T. Alfrey, Jr.
Polymer Research Lab.
Dow Chemical Company
Midland, MI 48640

1 Prof. N. Brown
1 Metallurgy Dept.
Univ. of Pennsylvania
Philadelphia, PA 19104
Dr. S. Krimm  
Univ. of Michigan  
Ann Arbor, MI 48104  

R. S. Marvin  
National Bureau of Standards  
Washington, DC 20234  

1 Prof. John F. Fellers  
Polymer Science & Engineering Program  
419 Dougherty Engineering Building  
1 University of Tennessee  
Knoxville, TN 37916