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AUTHORITY

COLUMBNIAN ALLOYS

The parameters required to form beaded heatshield panels of the CB-752 and WC-129Y alloys in a thickness of 0.012 inch were described in two reports by General Dynamics (1,2). The evaluations included (1) a fairly extensive series of roomtemperature tensile tests to determine the effects of tensile strain- and stress-relieving treatments on the uniform percent elongation of both alloys and (2) the development of a cooling and ambient-room-temperature fabrication methods to form the panels desired.

The strain tests showed both alloys workhardened appreciably after tensile prestraining at levels of 8 to 10 percent elongation. Stress relieving these materials after 10.5 to 12 percent prestrain increased the available ductility for forming.

Beaded panels of both alloys were readily formed using hydropress equipment and the Verson-Wheelon press. Female die forming was necessary to avoid buckling of the 0.012-inch-thick panels in the initial forming stages. Additional forming stages continued in male dies. Most results were obtained by interposing the workpiece between a 0.012-inch-thick caul sheet of Type 321 stainless steel and the forming force. Teflon films of 2-mil thickness were found effective as a die lubricant. A "picture-frame" type of pressure plate, over the panel border, with the opening close to the bead configurations, was regarded as a key factor in the prevention of buckles during forming.

The finished columbium alloy panels were about 12.4 by 15.7 inches in size, and each contained 9 parallel bead corrugations about 0.9 inch wide by 10.5 inches long. Bead heights of about 0.31 inch were formed in the single annealed CB-752 sheet without stress relief between forming stages. The maximum forming stretch was 12 percent in 1 inch. The WC-129Y panels were formed with bead heights in the range of 0.35 to 0.37 inch after stress relief between forming stages. The maximum stretch was 17 percent in 1 inch.

A single panel of the WC-103 columbium alloy was also formed using the same procedures. This panel, which had bead heights from 0.33 to 0.38 inch, was formed with greater ease than with either the CB-752 or WC-129Y alloys. Thus, with WC-103 material, beads with 12 percent maximum stretch in the female die were possible to form. Whereas 8 to 10 percent stretch was observed in this die with the other alloys. However, excessive "orange peel" was observed in the WC-103 material at the highest points of strain, indicating the desirability of stress relieving prior to a 12 percent stretch.

In continuing studies of columbium alloys at the Bureau of Mines, columbium-hafnium-tungsten alloys containing up to 10 atom percent boron were investigated. In columbium-hafnium alloys, up to 5 atom percent boron additions appeared to benefit fabricability. Likewise, in ternary columbium-hafnium-tungsten alloys, alloys containing 2 atom percent boron were apparently more fabricable than boron-free alloys, but higher boron contents impaired fabricability. In the annealed condition (1 hour at 1200 C or 2190 F), boron did not contribute to hot strength and quite definitely degraded oxidation resistance as evaluated by weight change.

COLUMBNIAN AND TANTALUM ALLOYS

Studies are in progress at Oak Ridge National Laboratory to strengthen refractory alloys at high temperatures through the precipitation of stable phases. Initially, the room-temperature precipitation hardening associated with eutectoid transformations in the tantalum-hafnium (eutectoid temperature of 1500 C or 2890 F) and columbium-titanium (eutectoid temperature of 610 C or 1132 F) systems was determined on samples of the Ta-5OHf and CB-58Zr (weight percent) alloys, respectively. Appreciable hardening in the Ta-50Hf alloy was indicated. Thus, the hardness of the solution-treated alloy increased from a value of 460 DPH to a peak hardness of about 790 DPH on aging at 850 C (1560 F) for about 10 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes. For the CB-58Zr alloy, only small hardness increases (about 20 DPH) were associated with aging at 850 C (1560 F) for 500 minutes.
1650 C but stronger at 2200 C. Optical and electron metallography showed that voids form at grain boundaries in both materials. In the PM material, grain boundaries appear to be immobile, and massive growth of voids initiates fracture with very little deformation. In the AM material, grain boundaries appear to be mobile and fracture seems to be forestalled by their ability to move. The relative creep rates can be explained on the basis of void growth and grain size.

In a series of studies at Lewis Research Center in which turbine-bucket applications were simulated, tungsten-alloy fibers were incorporated in Ni-25W-15Cr-2Al-2Ti nickel-base alloy by a slip casting-isostatic hot pressing procedure and the resultant structures were evaluated for stress-rupture strength and oxidation and impact resistance. The following alloys and wire diameters were investigated: W-2ThO$_2$ (0.015 inch and 0.010 inch), W-3Mo-2ThO$_2$, and W-1ThO$_2$ (both 0.020 inch) and 218 CS Ni (0.015 inch). The W-3Mo-2ThO$_2$ wire had the greatest tensile strength of all wires at 2000 F and was comparable to W-2ThO$_2$ at 2200 F. In stress-rupture tests both diameters of W-2ThO$_2$ wire proved to be strongest. Subsequent 100-hour stress-rupture tests at 2000 F in which 70 volume percent of wires were incorporated in the superalloy matrix yielded favorably high strength levels, e.g., 40 ksi as compared with 11.5 ksi for the best cast nickel alloys. Standard nickel-base alloys used for turbine-bucket applications have a stress-density value of about 60,000 inches for a 1000-hour rupture life at 1800 F. As illustrated in Fig. 1, which takes into account their densities, the nickel alloy W-2ThO$_2$-wire combination yielded a 200 F advantage at 1000 hours. The impact strengths as gaged by preliminary tests were comparable to those of cast superalloys. Exposed fiber ends apparently would need some sort of protection against oxidation in service environments.

In the final drawing stage, reductions in area as high as 99.7 percent were possible without intermediate annealing. Under these conditions, a room-temperature tensile strength of about 230 ksi was achieved in 10-mil-diameter wire, along with a yield strength of about 165 ksi and a ductility of 2 to 40 percent elongation were achieved by annealing between 600 and 800 C (1110 to 1470 F).

**CHROMIUM WIRE**

Improved procedures for the preparation of pure chromium wire have been described by workers at the Australian Aeronautical Research Laboratories. Briefly, these involve:

1. Canning the 10-pound, 3-inch-diameter arc-melted ingots in mild steel
2. Extrusion at 1200 C (2190 F) to 5/8-inch-diameter rod
3. Removal of the steel by machining to 1/2-inch-diameter rod
4. Swaging at 900 C (1650 F) to 0.220-inch diameter
5. Straightening and grinding to 0.190-inch diameter
6. Vacuum annealing at 950 C (1740 F)
7. Electropolishing and electroplating of nickel (0.1 to 0.2 mil), then copper (about 2 mils)
8. Vacuum annealing 1 hour at 750 C (1380 F)
9. Drawing, using a graphitic lubricant at 350 C (660 F)

In wide-ranging combinations of room-temperature strength (110 to 230 ksi) and ductility (2 to 40 percent elongation) were achieved by annealing between 600 and 800 C (1110 to 1470 F).

**REFRACTORY METALS, GENERAL**

Two more reports have been issued by Man-Labs on a comprehensive program aimed at characterizing the stability of a wide variety of refractory materials in air under high-velocity conditions. This program has included oxidation studies on refractory borides, graphites and JT composites, hypereutectic carbide-graphite composites, refractory metals (coated and uncoated), metal-oxide composites, and iridium-coated graphites. A principal goal of this program is elucidation of the relationship between hot gas/cold wall (HG/CW) and cold gas/hot wall (CG/HW) surface effects in terms of heat and mass-transfer rates at high temperatures.

One of these reports concerns a theoretical correlation of material performance with stream conditions. Here, published arc-plasma test data

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**FIGURE 1.** 100- AND 1000-HOUR SPECIFIC RUPTURE STRENGTHS OF CAST NICKEL ALLOY AND TUNGSTEN-THERIA WIRE-REINFORCED NICKEL ALLOY COMPOSITES TESTED AT 2000 F.
for refractory materials taken in eight different facilities were collected and examined by comparing the observed surface temperatures with calculated radiation-equilibrium values. Included are data for uncoated Hf-27Ta-2Mo, selected iridium alloys, and unalloyed tungsten (forged and as sintered) as well as tungsten which had been infiltrated with copper, zinc, and silver. Coated refractory-metal systems included are WSi,Ni, Sn-Al/7Ta-10W, and WSi/Ti. Wide variations in the ratio of calculated to observed temperature were encountered. Similar calculations performed for tests conducted in the present program yielded results close to unity, especially when melting is encountered. Larger 0.040-inch-thick sheet which will be used to clad resistant claddings for high-strength columbium and tantalum alloys. Thus, two alloys developed in earlier programs (i.e., Hf-24.5Ta-1.2Cr-0.6Si-0.12Mo and Hf-23.5Ta-1.8Cr-1.15Mo) are being prepared as 0.040-inch-thick sheet which will be used to clad substrates of the Cb-752 and Ta-10W alloys, respectively. The oxidation resistance of these composites at 2200 to 2700 F is to be determined as well as their mechanical properties after exposure at these temperatures. Secondary emphasis is being placed on the development of columbium-based alloys containing up to 50 weight percent hafnium with the objective of obtaining high-strength alloys having moderate to good oxidation resistance at 2200 to 2500 F.

The second of these reports was aimed toward the analytical formulation of a general surface-reaction description of the mass- and heat-transfer effects for a material body under hypersonic flight conditions.

In a study of potential thermionic materials, Gulf General Atomic determined interdiffusion rates in the binary-alloy systems of tungsten with tantalum, molybdenum, and columbium over the range of 2000 to 2500 C. In general, the diffusion rates were found to increase in the order of tantalum, molybdenum, and columbium, respectively. The vacuum work functions of the W-Ta, W-Mo, W-Cr, alloys at 1800 C (3270 F) were also determined as 4.39, 4.69, and 4.77 eV (electron volts), respectively.

Jetco-Omni-Skyscrapers has reported the results of an investigation of the phase relationships in two binary systems, tantalum-nitrogen and hafnium-nitrogen, and of the ternary tantalum-hafnium-nitrogen system. For isotropic sections from 1000 to 3200 C (1830 to 5790 F) are presented along with an isometric view of this ternary system from room temperature through 3200 C.

NEW PROGRAMS

Avco has issued its first progress report on an Army program exploring the feasibility of investment casting of columbium alloys. In this program, castings are being prepared by the AEM Metals Corporation using consumable electrodes of WC-20Ti and B-66 alloy compositions. These are skull melted and poured into molds in which initial tungsten-slurry dips provide the nonoxidic surface; a backup layer of zirconia is added as the final step in mold manufacture. Thirty-four molds are scheduled in the program to optimize the shell design, mold preheat temperatures, and to produce usable castings for the properties-testing phase. The mold analytical forms used to be included a fluidity test pattern, stress-rupture bars, oxidation-thermal fatigue paddles, and a lowing first-stage 155 turbine-nozzle vane. To date, threaded test-bar shapes of both alloys have been prepared which show acceptable surface quality and soundness.

Under a new Navy contract, the IIT Research Institute is expanding its previous work with oxidation-resistant hafnium alloys. Major emphasis is to be placed on the investigation of selected hafnium-tantalum base alloys as oxidation-resistant claddings for high-strength columbium and tantalum alloys. Thus, two alloys developed in earlier programs (i.e., Hf-24.5Ta-1.2Cr-0.6Si-0.12Al and Hf-23.5Ta-1.8Cr-1.15Al) are being prepared as 0.040-inch-thick sheet which will be used to clad substrates of the Cb-752 and Ta-10W alloys, respectively. The oxidation resistance of these composites at 2200 to 2700 F is to be determined as well as their mechanical properties after exposure at these temperatures. Secondary emphasis is being placed on the development of columbium-based alloys containing up to 50 weight percent hafnium with the objective of obtaining high-strength alloys having moderate to good oxidation resistance at 2200 to 2500 F.

REFERENCES


(13) Preliminary information from IIT Research Institute, Chicago, Ill., on U.S. Navy Contract N00019-70-C-0120.

DMIC Reviews of Recent Developments present brief summaries of information which has become available to DMIC in the preceding period (usually 3 months), in each of several categories. DMIC does not intend that these reviews be made a part of the permanent technical literature. Copies of referenced reports are not available from DMIC; most can be obtained from the Defense Documentation Center, Cameron Station, Alexandria, Virginia 22314.

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