Technical Memorandum

SEAWATER BATTERY EXPERIMENTS

by J. P. Wagner
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In the second phase, injection of nonreacting (nitrogen) and reacting (oxygen and chlorine) gases through the porous cathodes was investigated. Nitrogen had only a slight effect on increasing cell performance. With oxygen, noticeable improvements were obtained. With chlorine, the increases were dramatic.

Battery reaction mechanisms are discussed from an inductive viewpoint based on the Nernst equation and also from the theory of charge transport in ionic solutions.

The feasibility of employing the chlorine seawater battery for underwater applications is discussed.
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Technical Memorandum

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by J. P. WAGNER

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ABSTRACT

An experimental study pertaining to the development of a high energy density seawater battery is presented. The anode was a technical grade magnesium alloy, AZ31B-O (95.8 wt. % Mg), while the cathodes were sintered porous stainless steel with grain sizes of 165 micron, 35 micron, and 5 micron. In the initial phase of the study, cell orientation, cell geometry, electrode structure, and salinity effects were examined as to their effects on cell performance: current versus time characteristics and electromotive force (EMF). The highest current and power densities obtained were $J = 4.8 \text{ A/ft}^2$ and $P_D = 2.06 \text{ W/ft}^2$ for vertically oriented electrodes. The polarized cell EMF varied from only 0.4 volt to 0.5 volt for the 165-micron cathode over the wide range of parameters examined. A moderate dependence on cathode structure was observed; an EMF of 0.75 volt was measured for the 5-micron sample.

In the second phase, injection of non-reacting (nitrogen) and reacting (oxygen and chlorine) gases through the porous cathodes was investigated. Nitrogen had only a slight effect on increasing cell performance. With oxygen, a noticeable improvement was obtained: the mean current and power densities were $J = 18.6 \text{ A/ft}^2$ and $P_D = 16.8 \text{ W/ft}^2$. With chlorine, dramatic increases in $J$ and $P_D$ were obtained. $J$ was as high as 177 A/ft$^2$ while $P_D$ was 257 W/ft$^2$. The maximum projected power density based on the mean interelectrode spacing of 0.098 in and assumed (conservative) thicknesses of 0.32 in for the anode and cathode was 19.2 kW/ft$^3$. For both oxygen and chlorine, the cell EMF following injection also increased; it varied from 0.8 volt to 1.50 volts. No increases were obtained with nitrogen.
Battery reaction mechanisms are discussed from an inductive viewpoint based on the Nernst equation and also from the theory of charge transport in ionic solutions.

The feasibility of employing the chlorine seawater battery for underwater applications is discussed.
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<td>a</td>
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<tr>
<td>A_a</td>
<td>anode area, $L^2$</td>
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<td>A_c</td>
<td>cross-sectional area of cathode, $L^2$</td>
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<td>C</td>
<td>total ionic concentration, $C = C_+ + C_-$, moles/L$^3$</td>
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<td>C^*</td>
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<td>diameter, $L$</td>
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<td>d_a</td>
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<td>d_c</td>
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<td>D</td>
<td>diffusion coefficient, $D_+ + D_-$, $L^2/t$</td>
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<td>diffusion coefficient, $D_+ - D_-$, $L^2/t$</td>
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<td>E</td>
<td>reversible cell voltage, volts</td>
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<td>E_o</td>
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<td>E^c</td>
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<td>Faraday constant, 96,500 C/eq</td>
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<td>F_C^o</td>
<td>total concentration of charge, C/L$^3$</td>
</tr>
<tr>
<td>h</td>
<td>charge transfer coefficient, cm/sec</td>
</tr>
<tr>
<td>I</td>
<td>current, mA</td>
</tr>
<tr>
<td>I_{ss}</td>
<td>steady-state or polarized cell current, mA</td>
</tr>
<tr>
<td>Symbol</td>
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<tr>
<td>$I_{0.886}$ or $I_{1.672}$</td>
<td>current for a given anode diameter (in), mA</td>
</tr>
<tr>
<td>$J$</td>
<td>current density, $A/L^2$</td>
</tr>
<tr>
<td>$K$</td>
<td>conductivity, $R^{-1}L^{-1}$</td>
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<td>$K_{sp}$</td>
<td>solubility product constant, dimensionless</td>
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<td>$K_D$</td>
<td>Darcy permeability, $L^2$</td>
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<td>$L$</td>
<td>interelectrode spacing, $L$</td>
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<td>$m$</td>
<td>electrode thickness, $L$</td>
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<tr>
<td>$\dot{m}$</td>
<td>mass flow rate, $M/t$</td>
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<td>$\dot{M}_a$</td>
<td>rate of dissolution of anode, $M/t$</td>
</tr>
<tr>
<td>$\dot{M}_c$</td>
<td>difference between rate of formation and removal of $H_2$ at the cathode, $M/t$</td>
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<tr>
<td>$Nu$</td>
<td>Nusselt number, dimensionless</td>
</tr>
<tr>
<td>$P_i$</td>
<td>$i$-th product</td>
</tr>
<tr>
<td>$P$</td>
<td>porosity, dimensionless</td>
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<td>$\Delta p$</td>
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<tr>
<td>$q$</td>
<td>charge density, $q = F(C_+ - C_-), C/L^3$</td>
</tr>
<tr>
<td>$r_i$</td>
<td>$i$-th reactant</td>
</tr>
<tr>
<td>$R$</td>
<td>gas constant</td>
</tr>
<tr>
<td>$t$</td>
<td>time</td>
</tr>
<tr>
<td>$T$</td>
<td>Kelvin or absolute temperature, $273 + ^\circ C$</td>
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\( u \) ionic mobility, \( L^2/V-t \)

\( \vec{v} \) vector flow velocity, \( L/t \)

\( v \) cross-sectional velocity, \( L/t \)

\( v_o \) superficial velocity, \( L/t \)

\( V_p \) pore volume, \( L^3 \)

\( V_t \) total volume (solids plus voids), \( L^3 \)

\( X_i \) weight fraction, dimensionless

\( \eta_+ \) transference number for a positive ion, dimensionless

\( \tau \) relaxation time, \( \tau = \varepsilon \sigma / K, t \)

\( \lambda \) film thickness, \( L \)

\( \chi \) hydrogen blockage factor, dimensionless

\( \rho \) fluid density, \( M/L^3 \)

\( \mu \) fluid viscosity, \( M/L-t \)
1. INTRODUCTION

The use of batteries as high energy, primary power sources for underwater applications, specifically propulsion, has been considered by a number of investigators at least as early as 1950 (Ref. 1). Batteries have several attractive features that are especially suitable to this type of environment: they are simple to operate (charge and recharge cycles), self-contained, portable, relatively independent of external conditions, have high conversion efficiencies ($\eta \approx 85\%$), and generally have excellent storage properties. Although significant advances have been made in recent battery development programs, the principal drawback is their low energy densities, e.g., 30-150 W-hr/lb (at $\eta = 85\%$) (Ref. 2) versus values of around 1450 W-hr/lb (at $\eta = 25\%$) for conventional thermodynamic fuels (Ref. 3). In the lower region of the energy density range are batteries of moderate activity, e.g., the Ag-Zn battery and the seawater battery (Mg anode-inert cathode), while a battery made from elements or compounds at opposite extremes in the electrochemical activity series, such as the Li-Cl$_2$ battery, is illustrative of the upper region.

One aspect of interest in the Underwater Studies Program at APL is the development of a high energy density battery for underwater applications. The battery is subject to several rigid, yet arbitrary, constraints:

1. Generate 35 to 50 kW of continuous power for periods of up to 1 hour.
2. Operate within a cylindrical configuration, 21 inches x 7 feet, i.e., 16.8 ft$^3$.
3. Be reliable and have good storage properties.
4. Be cost effective.
The initial choice for study is the seawater battery because of low cost, high reliability, long life, and relative independence of operating depth and salinity (Ref. 4). Also, no protective case is required for electrolyte storage since the electrolyte is on-site seawater. Since a fivefold increase in the power output of present-day seawater batteries is needed to satisfy the above constraints (Ref. 5), more advanced electrochemical power sources are also being considered. They are a Mg anode-porous graphite cathode fuel battery using flowing liquid iodine at the cathode (Ref. 6), high temperature batteries having molten anodes and cathodes (Ref. 7), and a Li-Cl₂ battery similar to the Li-Sn battery in Ref. 8.

The present study examines the effects of a systematic variation of experimental parameters on the performance characteristics of a seawater battery. Although the results of this study demonstrate the feasibility of employing a seawater battery using gaseous chlorine injected through porous cathodes for practical high energy density applications, further study is required in a number of areas. Cell performance characteristics are needed for commercially available porous magnesium and lithium composites. Provisions for recycling the injected gas from cell to cell should be investigated, as well as other less toxic reacting gases or gas mixtures. Before scale-up of a small laboratory cell to a practical prototype much effort will be needed before the complex transport processes occurring within a seawater battery are adequately understood.
2. APPARATUS AND PROCEDURE

A schematic of the apparatus used in this study is given in Fig. 1. It consists of a cylindrical cell \((d = 2-1/8 \text{ in}, h = 3 \text{ in})\) of lucite and a measuring device to monitor current and voltage. Another cell made from a glass tube \((d = 2 \text{ in}, h = 4 \text{ in})\) and two rubber stoppers, one at each end of the tube, to hold the electrodes was used in the initial no-flow experiments. The lucite cell allowed one to measure accurately the interelectrode spacing and also prevented leakage of electrolyte during gaseous injection. The electrode holders were circular disks machined out of the inert plastics: Teflon, Nylon, and Lucite. The electrodes fit securely into the electrode holders for all non-flow tests. However, during the flow tests the cathodes were glued into the electrode holders in order to prevent gas leakage around the outer edge of the electrode and around the wire carrying the current to the measuring device.

Kintel microvolt ammeters Nos. 203R and 203A-R were used to measure current and voltage for all tests, except when the current exceeded 1 A, the upper limit of the ammeter. A standard Weston No. 430 ammeter sufficed for the higher amperages.

To ensure reproducibility and yet reuse the magnesium anode it was necessary to etch the magnesium before a run. A dilute solution of 3 N HCl was a satisfactory etching solution for the anode, as was a 2 N NaOH solution for the cathode for all runs in which a moderate reaction was observed. This occurred for values of \(P_D\) less than around \(P_D = 10 \text{ W/ft}^2\) and a solution \(pH \approx 7\). A concentrated solution of HCl was used to dissolve a dark crud-like material that covered the anode for \(P_D\) values greater than 10 W/ft²; the corresponding solution \(pH\) was 8.5. Because it was difficult to obtain reproducibility even after this foreign matter was removed, a new electrode was subsequently used whenever these conditions were observed. A white gel-like substance appeared on the cathode following the same vigorous reactions.
Fig. 1  SEAWATER TEST CELL

Fig. 2  CURRENT-TIME BEHAVIOR FOR TWO HORIZONTAL CELL ORIENTATIONS; \( t = 0.63 \) INCH, \( A_A / A_C = 0.0885 \)
Dilute 3 N HCl was used to dissolve this precipitate. For chlorine injection the stainless steel cathode was used only once since erosion occurred during testing. Following cleaning, both electrodes were rinsed with tap water and finally with copious quantities of distilled water. Current was then measured as a function of time.
3. EXPERIMENTAL RESULTS

CELL ORIENTATION AND GEOMETRICAL EFFECTS

Because of hydrogen evolution at the inert stainless steel cathode, the effects that free convection has on the current versus time characteristics were examined for three cell orientations: electrodes oriented horizontally (cathode lower and cathode upper), and electrodes aligned with gravity. In the paragraphs below a comparison is made between the performance characteristics for horizontal electrodes.

In Fig. 2 it is seen that an orientation having the cathode in the lower position leads to an increase in the current of about 33% (I = 0.63 in). For the smallest mean interelectrode spacing tested, 0.098 in, the results in Fig. 3 show that for the first 24 minutes the mean current increases with increasing Ac/Aa. In Fig. 4, after the initial transient decay period of about 8 minutes, it appears that the current for the Ac/Aa = 0.0865 curve will be greater than that for Ac/Aa = 11.6. Although the results shown in Figs. 3 and 4 do not demonstrate which orientation is best, additional data shown in Fig. 14 again favor the cathode in the lower position. Another interesting observation of the data in Figs. 3 and 4 is that the current never approaches a steady-state value but fluctuates between an ill-defined upper and lower limit over short time intervals. Visual observation indicated that this effect was due to hydrogen gas evolution in the form of a large bubble for each discrete increase in the current.

Additional results for interelectrode spacings of 4 = 0.315 in and 0.63 in presented in Figs. 5 and 6, respectively, show an orderly approach to a steady-state current with increasing time. Note also that the current extrapolated to t = 0 on semilog plots (not shown) shows an orderly increase with Ac/Aa. Another important observation is that the EMF for t > 25 min only varied between 0.41 to 0.50 volt for Ac/Aa = 1.00, 3.25, 11.6 and 16.5. Because of this observation the EMF was
Fig. 3 CURRENT-TIME BEHAVIOR FOR DIFFERENT CATHODE TO ANODE AREA RATIOS, $A_c/A_a$, AT A HORIZONTAL ORIENTATION; $r = 0.098$ INCH, $d_a = 0.492$ INCH

Fig. 4 CURRENT-TIME BEHAVIOR FOR TWO DIFFERENT AREA RATIOS, $A_c/A_a$, AT A HORIZONTAL ORIENTATION; $r = 0.098$ INCH
Fig. 5  CURRENT-TIME BEHAVIOR FOR DIFFERENT CATHODE TO ANODE AREA RATIOS, $A_c/A_a$, AT A HORIZONTAL ORIENTATION, $\theta = 0.315$ INCH, $d_e = 0.482$ INCH
Fig. 6  CURRENT-TIME BEHAVIOR FOR DIFFERENT CATHODE TO ANODE AREA RATIOS, \( \frac{A_c}{A_a} \), AT A HORIZONTAL ORIENTATION; \( \xi = 0.63 \) INCH, \( d_a = 0.492 \) INCH
measured following the initial transient current decay. Preliminary tests showed a slight decrease of the cell voltage with time; however, once the current was measured (for times as short as $t \approx 30$ sec) and then the cell voltage remeasured, a decrease in the EMF by around a factor of 2 was observed. Computation of the power output based on a polarized cell current and the corresponding EMF will, therefore, lead to a conservative value for the effective power output of a cell.

The current-time behavior shown in Fig. 7 is illustrative of the data obtained for vertically oriented electrodes at an interelectrode spacing of 0.098 in and large values of $\frac{A_c}{A_a}$ (here $\frac{A_c}{A_a} = 11.6$). The current fluctuates between a lower base level of about 4 mA and an upper limit of 6 mA after the decay of the transient current. Here the mean current is around 5 mA. This is a considerable improvement over the 2 to 3 mA obtained for the horizontally oriented electrodes of Figs. 3 and 4. An EMF of 0.42 volt at 64 minutes again lies within the EMF range obtained previously. Thus the cell's current density, power output, and power density are greater for vertically oriented electrodes than for horizontal orientations. Additional experimental evidence that supports this observation is given in Fig. 14. It is also clear that the steady-state power outputs of the cells are greatest for the smallest interelectrode spacing tested -- 0.098 in.

ANODE EFFECTS

In this section the effect of variation of anode area was examined for values of $\frac{A_c}{A_a} = 1.00, 3.25, \text{ and } 3.56$. Here the intent is to determine whether or not the current and the power densities are radially uniform. Or, in other words, is one justified in using values of current density and power density obtained on small lab-type cells to make estimates of overall power outputs on much larger cells (a scaling-up problem), when the mechanisms of mass transport and hydrogen bubble formation and evolution are, at best, incompletely understood.
Fig. 7  EFFECT OF HYDROGEN EVOLUTION ON THE CURRENT-TIME CHARACTERISTICS FOR A VERTICALLY ORIENTED CELL AT $A_e/A_g = 11.6$, $\delta = 0.098$ INCH
In Fig. 8 a comparison of the current-time characteristics is made for three different anode diameters: 0.492 in, 0.886 in, 1.672 in at $A_c/A_a = 1.00$ for horizontal orientation. In going from $d = 0.492$ in to $0.886$ in, the current following the initial transient decay increased by about tenfold, while the area only changed by a factor of 3.25. An increase in $d$ from $d = 0.886$ to $d = 1.672$ did not lead to a corresponding uniform increase in $I$; actually the $I$'s at $t = 12$ minutes are nearly equal. In Fig. 9 similar comparisons are made for a vertical cell orientation. A steady-state current of around 6.3 mA for the $d = 0.492$-in anode should correspond to a value of 20.5 mA at $d = 0.886$; this is reasonably close to the observed value of 18.5 at $d = 0.886$ in. In going to $d = 1.672$ in, one would expect a value of $I_{1.672} \approx 3.56 I_{0.886}$ or 61.9 mA. Values of 43 mA and 47 mA obtained on two separate runs fall far short of this projected value. Data points are only plotted up to $t = 14$ minutes since these runs were then subjected to gaseous injection and are plotted in Figs. 17 and 18. Further comparison in Fig. 10 for a $d = 0.492$-in anode at $A_c/A_a = 3.25$ with a $d = 0.886$-in anode at $A_c/A_a = 3.56$ is in close agreement since $I_{3.56}$ is expected to be roughly equal to $I_{3.56} = 3.25 I_{3.25}$, where $I_{3.25} = 5.5$ mA. That is, the projected value of 17.8 mA compares favorably with the observed value of 21 mA. It is again interesting to note that the EMFs only vary from 0.40 volt to 0.45 volt for the two orientations and the range of geometries over time intervals from 16 minutes up to 37 minutes. From a design viewpoint, the results presented in Figs. 8 - 10, inclusive, indicate that a cell made up of a series of small electrodes would have higher current and power densities than one made up of a single electrode of equivalent area. Or, alternatively, that the current density does not appear to be radially uniform.
Fig. 3 CURRENT-TIME DEPENDENCE ON ANODE AREA FOR A HORIZONTAL CELL ORIENTATION AT A<sub>N</sub>: A<sub>n</sub> = 1.00. ( = 0.096 INCH)
Fig. 9  CURRENT-TIME DEPENDENCE ON ANODE AREA FOR A VERTICAL CELL ORIENTATION: $A_2/A_1 = 1.26, (\approx 0.000$ INCH)

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Fig. 10 COMPARISON OF CURRENT-TIME BEHAVIOR FOR TWO DIFFERENT ANODE AREAS AT SELECTED AREA RATIOS, $A/A_e$ FOR A VERTICAL CELL ORIENTATION; $\bar{z} = 0.088$ INCH
CATHODE STRUCTURE

Porous stainless steel (316L) cathodes of three different grain sizes -- 165μ, 35μ, and 5μ -- were examined as to their effect on the current-time behavior for two different cell orientations. In Fig. 11 one observes a somewhat similar overall behavior for all three grain sizes. Closer inspection reveals that the 165μ test has a higher lower trendline, lying between 2 mA and 2.8 mA, than either the 35μ or the 5μ, which form a baseline at about 0.5 mA. The upper data points for the 165μ and the 35μ runs are about equal, while the 35μ points are slightly lower. Even though the EMF for the 165μ run measured at t = 58 min is 16.7% and 33% lower than the EMF's at t = 31 min and t = 25 min for the 35μ and 5μ cathodes, respectively, these differences are not sufficient to increase the cell power output above that of the 165μ cell.

The data in Fig. 12 for 35% salinity clearly show that a 165μ cathode more effectively reduces cell polarization than the 35μ cathode. The 5μ cathode was not tested because it appeared that its pores were more prone to clogging by cathodic deposits than either the 35μ or the 165μ cathodes.

The porosities of the 165μ, 35μ, and 5μ samples are 60.5%, 46.5%, and 43%, respectively. From a structural viewpoint the 165μ cathode would again be preferred because it has the largest surface area (PAc) perpendicular to the direction of charge transport.

SALINITY EFFECTS

Artificial seawater representing a typical analysis of seawater (Ref. 9) was prepared by dissolving properly ratioed quantities of NaCl, Na2SO4, MgCl2, KCl, and CaSO4 in a measured

\[ \text{The cathodes in Figs. 2 - 10 and 13 - 20 were the 165μ grain size.} \]
Fig. 11 CURRENT-TIME BEHAVIOR FOR THREE DIFFERENT CATHODE PORE SIZES AND A HORIZONTAL CELL ORIENTATION; \( \varepsilon = 0.098 \) INCH, \( A_a/A_c = 11.6 \)
Fig. 12  CURRENT-TIME BEHAVIOR FOR TWO DIFFERENT CATHODE PORE SIZES AT 35% SALINITY AND VERTICAL CELL ORIENTATION; \( \epsilon = 0.698 \) INCH, \( A_p/A_e = 1.00 \)
volume of distilled water. Compounds containing Br\(^-\), \(\text{BO}_3\)\(^-\), and \(\text{PO}_4\)\(^2-\) were not added since these anions were present in concentrations not exceeding 67 ppm. The effects of salinity\(^\dagger\) on \(I\) versus \(t\) data are shown in Fig. 13 (see also Fig. 18 and page 23 for an explanation of the flagged symbols). Here \(C^*\) is a molal type concentration, i.e., a given amount of salt was added to a graduated cylinder and distilled water was added to make the volume equal to 1L. For the lower concentrations \(C\) may be used interchangeably with \(C^*\), since the final solution volume is essentially independent of small quantities of salt.

The data show a rather strong dependence of the current on salinity. For high salinities the current is not uniform with time. The non-uniform behavior is not identical to the type of behavior observed previously for the discrete hydrogen evolution in Figs. 3, 4, and 7. It appeared that chunks of reacted anode material (a dark crud-like substance) broke off the surface, thereby exposing a fresh surface that led to the increase in the current. The solution became cloudy and the \(pH\) changed from about 6.5 - 7.0 before the test to 5.5 after. A white precipitate also deposited on the stainless steel cathode. The dark crud appears to be a mixture of metallic hydroxides resulting from the primary and secondary cell reactions given in Ref. 10:

**Primary reaction**

<table>
<thead>
<tr>
<th>Anode</th>
<th>(\text{Mg} - \text{Mg}^{++} + 2e^-)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Cathode</td>
<td>(2\text{H}_2\text{O} + 2e^- \rightarrow 2\text{OH}^- + \text{H}_2)</td>
</tr>
</tbody>
</table>

\(E^0\) for these reactions:

\(\text{Mg} + 2\text{H}_2\text{O} \rightarrow \text{Mg(OH)}_2^+ + \text{H}_2^+ 1.542\)

\(\dagger\) For the \(I\) versus \(t\) data given in Figs. 1 - 11 and 14 - 17, the salinity is 7 g/l. The salinity is defined as grams salt/kg solution \(\times\) 100%.
Fig. 13 EFFECT OF CONCENTRATION OF ELECTROLYTE ON THE CURRENT-TIME CHARACTERISTICS FOR A VERTICALLY ORIENTED CELL; \( \varepsilon = 0.008 \) INCH, \( A_L/A_w = 1.00 \)
Secondary reaction

\[ \text{NaOH} + \text{MgCl}_2 \rightarrow 2\text{NaCl} + \text{Mg(OH)}_2 \]

obtained from

Anode \[ 2\text{Cl}^- + 2\text{Mg}^{++} \rightarrow \text{MgCl}_2 \]

Cathode \[ \text{OH}^- + \text{Na}^+ \rightarrow \text{NaOH} \]

The white precipitate is probably a mixture of calcium and aluminum hydroxides.

Higher salinities than the \( C^* = 112 \text{ g/l} \) were tested; however, the measured current dropped abruptly after a few minutes from the mA range to the pA range. Upon reducing the electrode distance to the point where the electrodes were touching, the current again returned to the mA range. In all cases the current was much less than that obtained for the \( C^* = 112 \text{ g/l} \) test. It is possible that at extremely high salt concentrations a cell is set up between the wire leading to the measuring device and the anode which results in shorting out the main cell reaction. No effort was made to explore possible methods of circumventing this effect.

The strong dependence of cell reactivity on salinity is in contrast to the relative independence pointed out in Ref. 4. This apparent discrepancy is probably due to the use of a different type of a cathode (sintered porous stainless steel, type 316L) over a wider range of salinity. Chemical composition of the anode and the nature of its surface also enter into the problem. For example, pure magnesium will react in moist air (Refs. 11 and 12), whereas high salinity solutions are needed for appreciable reaction with the AZ31B-O magnesium alloy.

**GASEOUS INJECTION EFFECTS**

**Nonreactive Gases -- Nitrogen**

As previously observed, e.g., Figs. 3, 4, and 7, removal of \( \text{H}_2 \) gas from the electrode
surfaces via a free-convective process leads to a
discrete increase in the current. In this second
phase of the study, the effects of gaseous injection
through the porous cathodes (a forced convective
process) on the current-time behavior are examined
for nonreactive and reactive gases. Here nonreactive
means that the foreign gas cannot undergo a reaction
with the seawater reaction scheme given on page 20
while the contrary holds for a reactive gas. A
discussion of possible mechanisms whereby the two
reactive gases can undergo chemical reaction is
given in Section 6 on the seawater reaction
mechanism.

In Fig. 14 a comparison is made between
the effects of $N_2$ injection on the current-time
characteristics for three cell orientations at
$A_c/A_a = 1.00$. The solid vertical line drawn through
the bottom of any given symbol means that gas flow
was started immediately after that data point was
recorded; the two solid vertical lines around 5 mA
in Fig. 14 indicate that gas flow occurred at that
corresponding time on the abscissa but that data
points were not taken at those times. A single
vertical line followed by ordinary symbols means
that gas injection consisted of a pressure pulse
generally lasting a few seconds (never more than
around 10 sec); two or more flagged symbols mean
that the injection was operated in a continuous
mode over that time interval. An estimate of the
amount of gas injected is given in the Appendix.

For the vertically oriented electrodes
a slight increase in the current following injection
from 4.9 mA to 5.3 mA is noted. For the horizontal
case, in which the cathode was the lower electrode,
a near doubling of the current took place from
2.3 mA to 4.5 mA after the first $N_2$ pulse and
then from 2.1 mA to 3.5 mA after the second pulse.
With the anode in the lower position the current
increased slightly from 2.5 mA to 2.8 mA after
pulsing. Since the EMF's were not affected by $N_2$
injection, that is, they are all once again around
0.4 - 0.5 volt, the vertically oriented cell has
the highest power output. Also, the cell with the
anode in the lower position (horizontal orientation)
produced the lowest power output.
Fig. 14 COMPARISON OF THE EFFECTS OF NITROGEN INJECTION ON THE CURRENT-TIME CHARACTERISTICS FOR THREE CELL ORIENTATIONS; $A_c/A_d = 1.00, \ell = 0.098$ INCH

$E_{10} = 0.44$ V

$E_{22} = 0.41$ V

$E_{15} = 0.40$ V

$d_a = 0.492$ in.

$d_c = 0.492$ in.

Fig. 15 EFFECT OF NITROGEN INJECTION ON THE CURRENT-TIME CHARACTERISTICS FOR TWO DIFFERENT CELL ORIENTATIONS; $A_c/A_d = 11.6, \ell = 0.098$ INCH
In Fig. 15 the two best performing cells are compared for dissimilar electrode areas, $A_c/A_a = 11.6$. Once again, the current output is greater for the vertically oriented cell. Another important observation is that operating the cell in a continuous $N_2$ injection mode from $t = 7.5$ min to $t = 22$ min did not lead to an increase in the current. Actually, any appreciable injection rates caused the current to fluctuate between upper and lower limits; note the data at $t = 5, 13,$ and 22 min. Only after continuous injection was stopped and the cell operated in the pulse mode, e.g., at $t = 22, 29, 37,$ and 43 min, etc., did the current increase from its baseline between 5 mA and 6 mA up to around 8 mA. Similar behavior was also observed for the horizontal orientation although the magnitudes differed.

In Fig. 16 the effect of $N_2$ injection on change in anode area at $A_c/A_a = 1.00$ is examined for horizontal orientation. The $d = 1.672$-in diameter anode run is the same as the shortened run in Fig. 8; the two remaining runs were repeats of the conditions tested in Fig. 8. The results shown in Fig. 16 in comparison with those of Fig. 8 show that the effect of forced convection gas injection on augmenting the current is greater than that of free or natural convection. This is easiest to see by comparing the 0.886-in and 1.672-in diameter runs in these two figures. For example, at $t = 6$ min (Fig. 16) the current went from 9 mA to 15 mA following injection, whereas the high-low for the corresponding case fluctuated between 9 mA and 11 mA for $t = 5$ min to $t = 20$ min. The current fell below 9 mA to a lower level of 6 mA following $N_2$ gas injection; however, this only occurred for a period of about 2 min where a second $N_2$ pulse increased its value to 13 mA.

For the $d = 1.672$-in anode (Fig. 16) the current at $t = 16$ min increased from about 7 mA to 17 mA following $N_2$ injection, whereas at $t = 21$ min the effect of natural convection resulted in an increase of I from 10 mA to 17 mA. An $N_2$ pulse later increased I from 17 mA to 21 mA. Also, at $t = 29$ min, an $N_2$ pulse raised the current from 10 mA to 19 mA.
Fig. 16  EFFECT OF NITROGEN INJECTION ON THE CURRENT-TIME CHARACTERISTICS FOR THREE DIFFERENT ANODE AREAS AT $A_A/A_a = 1.00$ AND $\ell = 0.098$ INCH
A true comparison of the data for the d = 0.492-in anode in Figs. 16 and 8 cannot actually be made. This is because of an order of magnitude difference in the steady-state currents at \( t = 10 \) min and the first appearance of natural convection at \( t = 16 \) min, where \( I \) increased from 0.82 mA to only 1.1 mA. However, at \( t = 20 \) min the current increased from 0.30 mA to 1 mA, a 233% increase which is much greater than the percentage increase in Fig. 16 in going from 2 mA to 4 mA. Close inspection of the experimental conditions revealed that the 0.492-in anode in Fig. 8 was etched about ten times in HCl, whereas the anode in Fig. 16 was etched not more than four times. Since it is well known in seawater battery studies that the chemical nature of Mg anodes as well as the level of impurities can alter significantly the performance characteristics (Ref. 13), the above explanation appears to be the primary reason for these discrepancies.

It is interesting to note that the cell EMF was independent of the mode and amount of \( N_2 \) injected through the anode. For the runs in Figs. 14, 15, and 16 the EMF only varied between 0.40 - 0.48 volt.

Reactive Gases -- Oxygen and Chlorine

In the preceding section it was seen that the effects of \( N_2 \) injection were to depolarize the cell via \( H_2 \) removal, thereby leading to an increase in the current. In these former tests the current increase was such that it was generally well below the current at \( t = 0 \) min; for the few cases in which it was not, it was only nominally above the value at \( t = 0 \) min. The purpose of the present section is to examine the effects of injection of reacting gases -- oxygen and chlorine -- on cell behavior.

In Fig. 17 \( N_2 \) and \( O_2 \) injections are compared for vertical electrodes, and the effect of \( O_2 \) injection on horizontal orientation (cathode lower) is also examined. For the vertical orientation the currents before injection agree to better
Fig. 17 CURRENT-TIME CHARACTERISTICS FOR NITROGEN VERSUS OXYGEN INJECTION IN TWO CELL ORIENTATIONS; $A_e/A_a = 1.00$, $\ell = 0.098$ in.
than 10%; however, following N₂ injection at t = 14 min I increased from 42 mA to only 45 mA, whereas for O₂ injection I went from 46 mA at t = 13 min to 97 mA at t = 20 min (note that injection was continuous). At t = 20 min the O₂ flow was stopped, and cell polarization then reduced the current over the next 5 min to 51 mA. At t = 25 min the O₂ flow was again started and this led to a current of 100 mA at t = 29 min. The EMF for the cell at t = 29 min was 1 volt, with and in the absence of flow.

For the horizontal orientation, injection of O₂ at t = 6 min increased the current from 51 mA to 63 mA. The value of 63 mA is in close agreement with the value one obtains by linear extrapolation of I to t = 0. Again the voltage measured was 1 volt (no flow).

In Fig. 18 the effects of O₂ injection in solutions of C*/7 g/l, 42 g/l, and 112 g/l are examined for vertical orientation for the dₐ = 1.672-in cell at Aᵥ/Aₐ = 1.00. Note also the data given in Fig. 13 for the d = 0.492-in anode, Aᵥ/Aₐ = 1.00. The C*/7 g/l test has been discussed in the previous figure; it is only included for comparative purposes. For the C*/112 g/l, two individual curves were obtained; the one with the Y symbol was obtained on an old Mg electrode that had undergone a vigorous reaction. The A symbol curve was obtained with a freshly etched Mg electrode. A discrepancy of this magnitude was previously observed with frequently etched electrodes (see the second paragraph of this section). For these curves a moderate (~20%) increase is noted in the current following injection in the continuous or pulse-type modes. However, the baseline current is much higher than that obtained even after injection for the C*/7 g/l test. The current for the C*/42 g/l curve following injection was measured incorrectly and thus is not included in the plot. In addition to noting the strong dependence of I versus t on concentration, it is important to again note that the EMF in these tests varied only from 0.87 volt to 1 volt.
Fig. 18  EFFECT OF OXYGEN INJECTION AND SALINITY ON THE CURRENT-TIME CHARACTERISTICS FOR A VERTICALLY ORIENTED CELL AT $A_s/A_a = 1.00$, $\ell = 0.098$ INCH
In Fig. 19 the current-time behavior for chlorine injection is given for vertical orientation for two cells where \( d_a = 1.672 \) in and \( d_a = 0.492 \) in, both at \( A_c/A_a = 1.00 \). For the \( d_a = 1.672 \)-in anode the cell was pulsed for approximately 2 sec at \( t \) slightly less than 11 min. This led to a rather abrupt increase in the current from 310 mA to 510 mA. At \( t = 12 \) min a \( \text{Cl}_2 \) pulse of about 1 sec again increased the current to 500 mA. A reading was not obtained during the 1-min interval between these pulses. However, when the \( \text{Cl}_2 \) injection was stopped, at \( t = 12 \) min, the current decreased to its baseline value of about 300 mA. Continuous injection of \( \text{Cl}_2 \) at \( t = 29 \) min where \( I = 210 \) mA for 1 min led to a dramatic increase in current to 900 mA. The current decreased in 3 min to its steady-state value of 200 mA. The cell EMF (without injection) was equal to 1.25 volts. One should also note that \( I \) before injection is equal to the value in Fig. 18 for the \( \Delta \) symbol run. Both tests were made with freshly etched anodes.

For the \( d = 0.492 \)-in anode another dramatic increase in current occurred from \( I = 29 \) mA to a maximum of \( I = 90 \) mA. The cell EMF at \( t = 19 \) min was 0.82 volt (no flow), while at \( t = 20 \) it was 1.05 volts. Allowing the cell to remain an additional 13 min without injection of any type a value of 0.47 volt was obtained. Thus it is seen that the effect of addition of small amounts of \( \text{Cl}_2 \) or oxygen is to increase both the current and the EMF via a chemical reaction. When the reaction goes to completion the behavior typical of the cell without gaseous injection is obtained. For continuous injection over a 10-min interval at \( C^e = 112 \) g/l the current increased from 1 A to 2.35 A while the EMF was 1.5 volts. This amounts to values of \( J, P_D, \) and \( P_{PV} \) of \( J = 154 \) A/ft\(^2\), \( P_D = 231 \) W/ft\(^3\), and 17.3 kW/ft\(^3\). For \( C^e = 25 \) g/l, i.e., a salinity representative of actual seawater, and a similar chlorine injection period the maximum current was around 2.7 A and the EMF was 1.45 volts. This gives values of \( J = 177 \) A/ft\(^2\), \( P_D = 257 \) W/ft\(^3\), and \( P_{PV} = 19.2 \) kW/ft\(^3\), which are slightly greater than the values for \( C^e = 112 \) g/l (\( J = 154 \) A/ft\(^2\), \( P_D = 231 \) W/ft\(^3\), \( P_{PV} = 17.3 \) kW/ft\(^3\)). This latter test shows that salinity is not an important
Fig. 19  CURRENT-TIME CHARACTERISTICS FOR CHLORINE INJECTION AND VERTICAL CELL ORIENTATION AT $A_c/A_v = 1.00$, $\varepsilon = 0.003$ INCH
parameter for chlorine injection. Here part of the cathode eroded while a rust colored compound deposited on the anode. The pH's for the solution bulk and for the liquid on the anode were 2 and approximately 4, respectively.

In Fig. 20 the effect of an external resistance on the current-time characteristics is examined. Resistances of \( R = 200\Omega, 1,125\Omega, 5,000\Omega, \) and \( 12,000\Omega \) were connected in line with the anode. An orderly decrease in current with increasing resistance is observed, and, upon removing the load, an abrupt increase in the current takes place. For the 12K resistor, the current increased from 50\( \mu \)A to 125\( \mu \)A following \( \text{Cl}_2 \) injection. For the 200\( \Omega \) resistor, at \( t = 38\frac{1}{2} \) min to \( t = 44 \) min, the current varied from 6.7 mA to 6.4 mA, respectively. This is more than 2.1 times its value of 2.4 mA at \( t = 10\frac{1}{2} \) min without \( \text{Cl}_2 \) injection. At \( t = 48 \) min for \( R = 1125\Omega \), the injection of \( \text{Cl}_2 \) led to only a nominal increase in the current from 1.3 mA to 1.6 mA. This is apparently due to electrode deterioration with continuous \( \text{Cl}_2 \) injection as confirmed by inspection of the electrodes immediately after completion of the test.
Fig. 20  CURRENT-TIME CHARACTERISTICS FOR CHLORINE INJECTION
AT DIFFERENT EXTERNAL RESISTANCES FOR VERTICAL CELL
ORIENTATION; \( A_e/A_d = 1.00, \ell = 0.098 \text{ INCH} \)
4. SUMMARY OF PERTINENT EXPERIMENTAL OBSERVATIONS

NONFLOW

1. Vertically oriented electrodes produce a greater power output than horizontal electrodes. The least favorable performance characteristics were obtained with the anode in the lower position—horizontal orientation.

2. The smallest interelectrode spacing tested, 0.098 in, produced higher overall power outputs than either the 0.315-in or the 0.63-in spacings.

3. The current and power densities were not radially uniform for the orientations tested—vertical and horizontal (cathode lower).

4. Variations in salinity strongly influenced the current-time characteristics.

5. The EMF for a polarized cell varied between only 0.4 - 0.5 volt and was independent of \( A_C/A_A \), salinity, interelectrode spacing, cell power output, and time. A slight dependence on cathode pore size was observed.

GASEOUS INJECTION

Nonreactive Nitrogen

Nitrogen had to be injected in a pulse-type manner (on-off) in order to increase the baseline current. An increase in the polarized cell current up to about 100% was obtained on the horizontally oriented cell (cathode lower); increases less than 12% were obtained for the vertical and the anode lower arrangements. The cell EMF, varying between 0.40 - 0.48 volt, was independent of the mode and amount of \( N_2 \) injected.
Reactive -- Oxygen and Chlorine

Continuous O₂ injection led to a 100% increase in the current over its estimated value at t = 0 for the vertical orientation. The increase was salinity-dependent. For the horizontal orientation the overall current increase was modest. The cell EMF following O₂ injection (with or without flow) was around 1 volt.

Dramatic increases in current were obtained for either a pulse- or continuous-type mode of Cl₂ injection for the vertical orientation. For the pulse-type mode the current increased by about 70% over its apparent value at t = 0, whereas it went up about 7000% for continuous injection. The cell voltage varied from 0.8 volt to 1.50 volts for a d = 1.672-in node (A_c/A_a = 1.00).
5. CELL PERFORMANCE

In this section comparisons are made between cell currents and power densities for a selected number of runs. A calculation of the minimum cell volume required to provide 35 kW of power for Cl₂ injection is also included.

In Tables 1 and 2 are computed values of current density, \( J \) (A/ft²), power density (W/ft²), \( P_D \), and maximum power density based on cell volume, \( P_{DV} \) (W/ft³), for horizontal and vertical orientations, respectively. The brackets (min) and (mA) indicate the time at which the voltage was measured and the value of current used to compute the current density. Although the actual thicknesses of the Mg anodes were 1/16 in and the porous cathodes 1/8 in, a value of 1/32 in was used in computing \( P_{DV} \). A negligible amount of Mg was used during any run while the cathode is inert and undergoes no erosion. Even if 2 A were drawn for 1 hour, Faraday's law requires that 0.0745 g equivalents of Mg or 0.906 g Mg undergo reaction. For a slice of 1/32-in thickness of the Mg would erode \( \rho \text{AZ31B-O} = 0.064 \text{ lb/in}^3 \). Thus, a 1/32-in thick Mg sample is a very conservative anode thickness.

The current measured covered a very wide range -- from 0.32 mA to 2700 mA, while the voltages ranged between 0.4 volt and 1.50 volts. Current densities and power densities did not exceed 4.8 A/ft² and 3.8 W/ft², respectively, even with \( O_2 \) or \( N_2 \) injection when \( C^* \leq 7 \text{ g/l} \). These values are much lower than the maximum values reported in Ref. 10 where \( J = 18 \text{ A/ft}^2 \) and \( P_D = 8.1 \text{ W/ft}^2 \) for the AZ31B-O alloy and a palladium and nickel coated screen (double 8 mesh) cathode. For \( C^* = 42 \text{ g/l} \) and \( O_2 \) injection \( J = 11.4 \text{ A/ft}^2 \) while \( P_D = 9.9 \text{ W/ft}^2 \). The highest values of \( J \) and \( P_D \) for \( O_2 \) injection were obtained when \( C^* = 122 \text{ g/l} \) \( J = 18.6 \text{ A/ft}^2 \) and \( P_D = 16.8 \text{ W/ft}^2 \). Dramatic increases in \( J \) and \( P_D \) were obtained with the Cl₂ injection. For the \( d = 1.672 \text{-in anode, } J = 17.7 \text{ A/ft}^2 \) and \( P_D = 257 \text{ W/ft}^2 \); while, for the \( d = 0.492 \text{-in anode, } J = 68.5 \text{ A/ft}^2 \) and \( P_D \) was as high as 72 W/ft².
<table>
<thead>
<tr>
<th>Diameter of Anode (in)</th>
<th>Diameter of Cathode (in)</th>
<th>A_c/A_a</th>
<th>Cell Spacing (in)</th>
<th>C* (g/l)</th>
<th>Injection Type</th>
<th>Current Range (mA)</th>
<th>EMF Volts (min)</th>
<th>A/ft² (mA)</th>
<th>P_D* (W/ft²)</th>
<th>P_DV** (W/ft²)</th>
</tr>
</thead>
<tbody>
<tr>
<td>0.492</td>
<td>0.492</td>
<td>1.00</td>
<td>0.63</td>
<td>7</td>
<td>--</td>
<td>1.7 - 4.1</td>
<td>0.43</td>
<td>1.52</td>
<td>0.655</td>
<td>10</td>
</tr>
<tr>
<td>Cathode Lower</td>
<td>0.492</td>
<td>0.492</td>
<td>1.00</td>
<td>0.315</td>
<td>7</td>
<td>0.94 - 4.7</td>
<td>0.42</td>
<td>1.14</td>
<td>0.49</td>
<td>16</td>
</tr>
<tr>
<td>1.672</td>
<td>1.672</td>
<td>1.00</td>
<td>0.098</td>
<td>7</td>
<td>N_p (Pulse)</td>
<td>7 - 42</td>
<td>0.40</td>
<td>0.92</td>
<td>0.37</td>
<td>28</td>
</tr>
<tr>
<td>1.672</td>
<td>1.672</td>
<td>1.00</td>
<td>0.098</td>
<td>7</td>
<td>N_p (Continuous)</td>
<td>48 - 63</td>
<td>1.00</td>
<td>3.8</td>
<td>3.8</td>
<td>285</td>
</tr>
<tr>
<td>Cathode Upper</td>
<td>1.672</td>
<td>0.492</td>
<td>0.0865</td>
<td>0.098</td>
<td>7</td>
<td>1.9 - 8.4</td>
<td>0.43</td>
<td>0.197</td>
<td>0.0842</td>
<td>6</td>
</tr>
<tr>
<td>0.492</td>
<td>0.492</td>
<td>1.00</td>
<td>0.098</td>
<td>7</td>
<td>N_p (Pulse)</td>
<td>2.3 - 2.8</td>
<td>0.44</td>
<td>1.9</td>
<td>0.835</td>
<td>63</td>
</tr>
</tbody>
</table>

*Based on anode area and the cell's steady state or a mean current from the I versus t plots.

**Assumes 1/32 in for the thicknesses of the anodes and cathodes; no allowance is made for the electrode holders.
Table 2
Performance Characteristics for Vertical Electrodes (Cathode -- 165μ Stainless Steel,
$A_c/A_e = 1.00$, $t = 0.028$ in)

<table>
<thead>
<tr>
<th>Diameter of Anode (in)</th>
<th>Diameter of Cathode (in)</th>
<th>Ce (g/l)</th>
<th>Injection Type</th>
<th>Current Range (mA)</th>
<th>EMF Volts (min)</th>
<th>$J^*$ (A/ft$^2$) (mA)</th>
<th>$P_D^*$ (W/ft$^2$) (W/ft$^2$)</th>
<th>$P_{DV}^{**}$</th>
</tr>
</thead>
<tbody>
<tr>
<td>0.492</td>
<td>0.492</td>
<td>0.035</td>
<td></td>
<td>0.32 - 0.40</td>
<td>0.49</td>
<td>0.267</td>
<td>0.13</td>
<td>10</td>
</tr>
<tr>
<td>0.886</td>
<td>0.886</td>
<td>0.35</td>
<td></td>
<td>3.8 - 4.1</td>
<td>0.42</td>
<td>0.915</td>
<td>0.384</td>
<td>29</td>
</tr>
<tr>
<td>0.492</td>
<td>0.492</td>
<td>7</td>
<td></td>
<td>5.9 - 6.7</td>
<td>0.41</td>
<td>4.8</td>
<td>2.06</td>
<td>154</td>
</tr>
<tr>
<td>0.492</td>
<td>0.492</td>
<td>7</td>
<td>$N_2$</td>
<td>4.9 - 5.3</td>
<td>0.40</td>
<td>3.96</td>
<td>1.59</td>
<td>119</td>
</tr>
<tr>
<td>0.492</td>
<td>0.492</td>
<td>42</td>
<td>$O_2$</td>
<td>10 - 17</td>
<td>0.87</td>
<td>11.4</td>
<td>9.9</td>
<td>740</td>
</tr>
<tr>
<td>1.672</td>
<td>1.672</td>
<td>112</td>
<td>$O_2$ (Continous)</td>
<td>210 - 370</td>
<td>0.90</td>
<td>16.6</td>
<td>16.8</td>
<td>1260</td>
</tr>
<tr>
<td>1.672</td>
<td>1.672</td>
<td>25</td>
<td>$Cl_2$</td>
<td>170 - 2700</td>
<td>1.45</td>
<td>257</td>
<td>19.15</td>
<td></td>
</tr>
<tr>
<td>0.492</td>
<td>0.492</td>
<td>112</td>
<td>$Cl_2$</td>
<td>26 - 90</td>
<td>0.82</td>
<td>68.5</td>
<td>56</td>
<td>4200</td>
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<tr>
<td>1.672</td>
<td>1.672</td>
<td>112</td>
<td>$Cl_2$</td>
<td>300 - 2350</td>
<td>1.5</td>
<td>154</td>
<td>231</td>
<td>17300</td>
</tr>
</tbody>
</table>

*Based on anode area and the cell’s steady state or a mean current from the $I$ versus $t$ plots.

**Assumes 1/32 in for the thicknesses of the anodes and cathodes; no allowance is made for the electrode holders.

*Based on the maximum values of $I$. 
For chlorine injection, \( P_{D_{\text{max}}} \) was as high as 19.2 kW/ft\(^3\). Thus, the minimum electrode volume needed to produce 35 kW is only 1.83 ft\(^3\). In the Appendix, the amount of chlorine needed to provide 35 kW for 1 hour with \( A_c - A_a = 2.4 \text{ ft}^2 \) is 19.9 ft\(^3\). This is too high from our viewpoint. However, the amount of Cl\(_2\) that reacted chemically is believed to be much less than this amount and, therefore, a practical cell could still be developed by recycling any unreacted chlorine from cell to cell.
6. DISCUSSION

SEAWATER REACTION MECHANISM

Explanations are offered for the increased voltages following O₂ and N₂ injection based on the seawater reaction mechanism and application of the Nernst equation.

For a cell operating reversibly, the voltage, \( E \), is given by the Nernst equation:

\[
E = E^0 - \frac{RT}{\varphi F} \ln \frac{\left( \frac{a_{P1}^{n1}}{a_{P2}^{n2}} \right)^{\frac{n1}{n2}} \cdots \left( \frac{a_{Pn}^{n_n}}{a_{Pn}^{n_n}} \right)^{\frac{n_n}{n_n}}}{\left( \frac{a_{R1}^{m1}}{a_{R2}^{m2}} \right)^{\frac{m1}{m2}} \cdots \left( \frac{a_{Rm}^{m_m}}{a_{Rm}^{m_m}} \right)^{\frac{m_m}{m_m}}},
\]

where \( E^0 \) is the difference in the standard half-cell voltages, \( R \) is the gas constant, \( T \) the absolute temperature, \( \varphi \) the number of equivalents, \( F \) the Faraday constant, \( a_{Pn} \) the activity coefficient of the \( n \)-th product, \( a_{Rm} \) the activity coefficient of the \( r \)-th reactant, and \( n_n \) and \( m_m \) are the stoichiometric coefficients. For pure Mg\( ^m \) the difference in the half-cell voltages for the seawater battery is \( E^0 = 1.542 \) volts. For the AZ31B-O Mg alloy an estimated value of \( E^0 \) is 1.48 volts (see Appendix).

For the cell reaction scheme

\[
\begin{align*}
\text{Mg} & \rightarrow \text{Mg}^{++} + 2e \\
\frac{2}{3} \text{Al} & \rightarrow \frac{2}{3} \text{Al}^{+++} + 2e \\
\text{Anode} & \\
\text{Mn} & \rightarrow \text{Mn}^{++} + 2e \\
\text{Zn} & \rightarrow \text{Zn}^{++} + 2e
\end{align*}
\]
Cathode \[ 8\text{H}_2\text{O} + 8\text{e}^- \rightarrow 8\text{OH}^- + 4\text{H}_2 \]

we obtain

\[
\text{Mg} \cdot \frac{2}{3} \text{Al} \cdot \text{Mn} \cdot \text{Zn} \cdot 8\text{H}_2\text{O} \rightarrow \text{Mg(OH)}_2 \cdot \frac{2}{3} \text{Al(OH)}_3 \cdot \text{Mn(OH)}_2 \cdot \text{Zn(OH)}_2 \cdot 4\text{H}_2
\]

for the overall cell reaction. The Nernst equation becomes

\[
E_{298} = 1.48 - \frac{0.0592}{2} \log \left( \frac{a_{\text{Mg(OH)}_2} \cdot a_{\text{Al(OH)}_3} \cdot a_{\text{Mn(OH)}_2} \cdot a_{\text{Zn(OH)}_2} \cdot a_{\text{H}_2}}{a_{\text{Mg}} \cdot a_{\text{Al}} \cdot a_{\text{Mn}} \cdot a_{\text{Zn}} \cdot a_{\text{H}_2\text{O}}} \right)^2
\]

where the number of equivalents transferred per metal is 2. For a cell voltage of \( E = 0.45 \) volt, which corresponds to the mean of the measured polarized cell voltages (for no flow and nitrogen injection), the Nernst equation requires that

\[
\frac{a_{\text{Mg(OH)}_2} \cdot a_{\text{H}_2}}{a_{\text{Mg}} \cdot a_{\text{H}_2\text{O}}} = 10^{35}
\]

Since the activity coefficient is proportional to concentration a value for the above ratio close to the order of \( 10^{35} \) follows from the solubility product constants for the metallic hydroxides which are (Ref. 14):

\[
\text{Mg(OH)}_2 \rightarrow \text{Mg}^{++} + 2\text{OH}^- \quad K_{sp} = 10^{-11}
\]
\[
\text{Mn(OH)}_2 \rightarrow \text{Mn}^{++} + 2\text{OH}^- \quad K_{sp} = 10^{-13}
\]
\[
\text{Zn(OH)}_2 \rightarrow \text{Zn}^{++} + 2\text{OH}^- \quad K_{sp} = 10^{-17}
\]
\[
\text{Al(OH)}_4^- \rightarrow \text{Al(OH)}_3^- + \text{OH}^- \quad K_{sp} = 2.5 \times 10^{-2}
\]
Aluminum can also precipitate out as the oxide Al₂O₃. Because aluminum is amphoteric, both strongly basic and acidic solutions will increase its solubility in solution. Its Kₚ value is subject to discussion since pH's of 8.5 were obtained at high reactivity. Incorporating the activities of the metals Mg, Al, Mn, and Zn, which are given by their respective mean weight fractions, 0.958, 0.03, 0.002, and 0.01, (see Appendix) into the ratio of the products of the activities gives

\[
\frac{a_{\text{Mg(OH)}_2} \cdot a_{\text{H}_2}}{a_{\text{Mg}} \cdot a_{\text{H}_2}} \approx 10^{3.5} a_{\text{H}_2}^4.
\]

Here \(a_{\text{H}_2O}\) was set equal to \(a_{\text{H}_2O} = 1.00\) in accordance with standard electrochemical conventions. The activity of H₂ remains an unknown; however, if \(a_{\text{H}_2}\) does not deviate too far from unity, then \(E_{298}\) would be equal to \(E_{298} \approx 0.45\). Without actually measuring the partial pressure of H₂ one cannot determine \(a_{\text{H}_2}\). For present purposes we are setting \(a_{\text{H}_2} = 1.00\) as required by the Nernst equation. There are additional reactions such as the secondary reaction,

\[
2\text{NaOH} + \text{MgCl}_2 \rightarrow 2\text{NaCl} + \text{Mg(OH)}_2,
\]

and the solution reactions,

\[
\begin{align*}
\text{CaSO}_4 & \rightarrow \text{Ca}^{++} + \text{SO}_4^- \quad K_{\text{sp}} = 1.95 \times 10^{-4} \\
\text{Ca}^{++} + \text{CO}_3^- & \rightarrow \text{CaCO}_3 \downarrow \quad K_{\text{sp}} = 10^8 \\
\text{Mg}^{++} + \text{CO}_3^- & \rightarrow \text{MgCO}_3 \downarrow \quad K_{\text{sp}} = 3.85 \times 10^4
\end{align*}
\]

where \(\text{CO}_3^-\) results from the solubility of ambient \(\text{CO}_2\) in \(\text{H}_2\text{O}\), which are of lesser importance and, thus, should not greatly affect the overall cell reaction in Eq. (2).
The addition of O₂ and Cl₂ through the porous cathode led to increases of the polarized cell voltage from the 0.4 - 0.5-volt range up to a range from around 1 volt to 1.50 volts. Taking E° = 1.00 and assuming that the overall cell reaction is still given by Eq. (2), i.e., at least for O₂ injection, we obtain a value of order 10⁻¹⁶ for the ratio of the products of the activity coefficients. This would require that the concentration of one or both products be lowered and/or the activity of the reactants be increased. Changes in the activities of the reactants leading to an order of magnitude change of 10⁻¹⁹ are unlikely; the volume of H₂O was actually observed to decrease slightly during a vigorous cell reaction, and a decrease in aMg would be offset by further dissolution of the Mg electrode. A direct reaction of O₂ with H₂ to form H₂O on the stainless steel cathode, thereby reducing aH₂, is unlikely because a two-body recombination reaction of gases must occur either on the stainless steel cathode (a noncatalyst) or in solution. Also, O₂ does not alter the solubility of the insoluble metal hydroxides. One remaining reaction scheme is for an additional electrochemical reaction to occur at the cathode, thereby raising the E° value while maintaining the primary reaction in Eq. (2). From the standard oxidation-reduction charts (e.g., Ref. 15) we obtain

\[
E^0 = \begin{align*}
O_2 + 2H_2O + 4e^- &\rightarrow 4OH^- & 0.40 \\
4H_2O + 4e^- &\rightarrow 4OH^- + 2H_2 & -0.828
\end{align*}
\]

Overall \( O_2 + 6H_2O + 8e^- \rightarrow 8OH^- + 2H_2 \) \(-0.428\) volt.

Cathode Reaction

Thus, the overall cell reaction is

Anode \( \sum_{i=1}^{3} w_i \cdot 2\frac{1}{3} Al - \sum_{i=1}^{3} w_i \cdot \frac{2}{3} Al^{+++} - 8e^- \) \( v = 2.30 \) V

Cathode \( O_2 + 6H_2O + 8e^- \rightarrow 8OH^- + 2H_2 \) \(-0.428\) V

Overall Cell Reaction \( \sum_{i=1}^{3} w_i \cdot \frac{2}{3} Al + O_2 + 6H_2O \rightarrow \sum_{i=1}^{3} w_i \cdot \frac{2}{3} Al^{+++} + 8OH^- + 2H_2 \) 1.882 volts
Here $M_i$ and $M_i^{++}$ are used to denote the metals Mg, Zn, Mn, and their ions, respectively. The Nernst equation for this two-step cathode reaction is

$$E_{298} = 1.882 - 0.0592 \log \left( \frac{a_{Mg(OH)_2} a_{Al(OH)_2} 2 a_{Zn(OH)_2} a_{H_2}}{a_{Mg} a_{Al} a_{Zn} a_{H_2O} a_{O_2}} \right).$$

with the exception of $a_{O_2}$, in the denominator of the ratio of activity coefficients and the $E^0$ value equal to $E^0 = 1.882$, identical to Eq. (2). For a value of $10^{35}$ for the ratio of activity coefficients $E_{298} = 0.85$, while for $10^{30}$ we obtain $E_{298} = 1.00$. This covers the range of the measured cell EMF's.

For $Cl_2$ injection a direct reaction to the chlorides is conceivable. This would lead to a reduction of the activity coefficients of the metal hydroxides and a corresponding increase in the cell EMF. A number of electrochemical reaction schemes are possible for $Cl_2$ injection; one that gives a reasonable net value for $E^0$ and also allows for soluble metal chloride formation is

\begin{align*}
Cl_2 + 2 e^- & \rightarrow 2 Cl^- \\
2Cl_2 + 2 e^- & \rightarrow 4Cl^- \\
2Cl^- & \rightarrow Cl_2 + 2 e^-
\end{align*}

For $Cl_2$ injection a direct reaction to the chlorides is conceivable. This would lead to a reduction of the activity coefficients of the metal hydroxides and a corresponding increase in the cell EMF. A number of electrochemical reaction schemes are possible for $Cl_2$ injection; one that gives a reasonable net value for $E^0$ and also allows for soluble metal chloride formation is

\begin{align*}
4Cl_2 + 8 e^- & \rightarrow 8Cl^- \\
2Cl_2 + 4 e^- & \rightarrow 4Cl^- \\
2Cl_2 + 4 e^- & \rightarrow 4Cl^- \\
4Cl^- & \rightarrow 4Cl_2 + 8 e^-
\end{align*}

Here $M_i$ and $M_i^{++}$ are used to denote the metals Mg, Zn, Mn, and their ions, respectively. The Nernst equation for this two-step cathode reaction is

$$E_{298} = 1.882 - 0.0592 \log \left( \frac{a_{Mg(OH)_2} a_{Al(OH)_2} 2 a_{Zn(OH)_2} a_{H_2}}{a_{Mg} a_{Al} a_{Zn} a_{H_2O} a_{O_2}} \right).$$

with the exception of $a_{O_2}$, in the denominator of the ratio of activity coefficients and the $E^0$ value equal to $E^0 = 1.882$, identical to Eq. (2). For a value of $10^{35}$ for the ratio of activity coefficients $E_{298} = 0.85$, while for $10^{30}$ we obtain $E_{298} = 1.00$. This covers the range of the measured cell EMF's.

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\begin{align*}
4Cl_2 + 8 e^- & \rightarrow 8Cl^- \\
2Cl_2 + 4 e^- & \rightarrow 4Cl^- \\
2Cl_2 + 4 e^- & \rightarrow 4Cl^- \\
4Cl^- & \rightarrow 4Cl_2 + 8 e^-
\end{align*}
Since the products of this reaction are soluble salts, the ratios of the activity coefficients cannot approach values of the order of $10^{45}$ for $E_{298} \sim 1$ volt. It appears likely therefore, that the primary cell reaction in Eq. (2) also occurs at reduced values for the activity coefficients of the metallic hydroxides. That is, the injection of Cl$_2$ gas leads to a partial dissolution of the insoluble hydroxides at a reduced value of $E_{298}$. The overall cell reaction, thus the cell EMF, is therefore a combination of the primary seawater reaction (Eq. (2)) and the complex reaction scheme (Eq. (4)).

**CHARGE TRANSPORT**

A fundamental approach to an understanding of the complex transport processes occurring in a seawater battery would require a solution to the multicomponent equations of continuity, motion, and charge transport subject to appropriate boundary conditions at the anode and cathode. Such a study would be of great value because the scale-up of a small laboratory-size battery to a practical prototype could be readily accomplished. Since this would require a great effort and is also not warranted at this time because the reaction mechanisms at the anode and cathode are not sufficiently well understood, a semiquantitative approach is adopted in order to outline possible alternative approaches.

If we assume the solution can be represented by a single component, uni-univalent electrolyte, then a solution to the time-dependent transport equation for charge (Ref. 16)

$$\frac{\partial q}{\partial t} = D \nabla^2 q - \frac{q}{\tau} - \nabla \cdot \mathbf{v} q + D' \nabla^2 C - \frac{D (\nabla C)^2}{q/F}$$

(5)

where

$$q = F(C_+ - C_-) \quad D' = D_+ - D_-/2 \quad C = C_+ + C_-$$

$$D = D_+ + D_-/2 \quad \tau = \varepsilon_0 / K$$
and integration over the electrode area would provide the current density-voltage versus time characteristics. Other standard definitions are for the conductivity, $K$, and the ionic mobility $u$:

$$K = u CF$$

$$u = \frac{DF}{RT}.$$ 

The terms in Eq. (5) from left to right are the diffusive, conductive, convective, and nonlinear fluxes, respectively. An order of magnitude comparison is made for these terms below.

The primary ions undergoing reaction in the seawater battery are $\text{Mg}^{++}$, $\text{OH}^-$, $\text{H}_2\text{O}^+$, $\text{Na}^+$, and $\text{Cl}^-$. The ionic mobilities for $\text{H}_2\text{O}^+$ and $\text{OH}^-$ are $3.64 \times 10^{-3}$ cm$^2$/volt-sec and $2.05 \times 10^{-3}$ cm$^2$/volt-sec, respectively, while the mobilities for $\text{Mg}^{++}$, $\text{Na}^+$, and $\text{Cl}^-$ are all about $5 \times 10^{-4}$ cm$^2$/volt-sec. The remaining ions $\text{SO}_4^{2-}$, $\text{K}^+$, $\text{Ca}^{++}$ from the compounds $\text{Na}_2\text{SO}_4$, $\text{KCl}$, and $\text{CaSO}_4$ are also in the above range of values. Thus a reasonable mean value of $u$ is $u = 1 \times 10^{-3}$ cm$^2$/volt-sec. Most ionic diffusion coefficients are about $10^{-5}$ cm$^2$/sec. For the highest salt concentration $C^* = 112$ g/l the total concentration of equivalents in solution is $2 \times 10^{-3}$ eq/cc. The conductivity from Eq. (6) is then $K \approx 0.2 \Omega^{-1}$cm$^{-1}$. A reference value for $q$ is $q = FC = (96500)(2 \times 10^{-3}) = 1.9 \times 10^{-2}$ C/cc. Now from Eq. (5) we obtain

$$DV^2q \approx Dq/L^2 \approx \frac{(10^{-5})(1.9 \times 10^2)}{(2.5 \times 10^{-1})^2} \approx 0(10^{-1})$$

$$q/\tau = \frac{K}{\varepsilon\varepsilon_0} q \approx \frac{(2 \times 10^{-1})(1.9 \times 10^2)}{(80)(8.85 \times 10^{-14})} \approx 0(10^{12})$$

$$v \cdot \nabla q \approx q/L = 0 \text{ (No Flow)}$$
The overwhelmingly large value for $\frac{q}{\tau}$ clearly indicates that ohmic conduction is the dominant factor. Now the flux of charge at the electrode surface is given by

$$J = -D\nabla q - D'F\nabla C - K\psi + \overline{vq} + j_a + j_c \quad (8)$$

where $j_a$ and $j_c$ refer to the fluxes due to the rate of dissolution of the Mg anode, and $j_c$ refers to the net difference between the rate of formation and rate of removal of $H_2$. An order of magnitude estimate for these electrode fluxes is:

$$D\nabla q \approx \frac{(10^{-5})(1.9x10^2)}{(2.5x10^{-1})} \approx (10^{-2})$$

$$D'F\nabla C \approx D'F \frac{C}{L} \frac{(5x10^{-6})(9.65x10^4)(2x10^{-3})}{(2.5x10^{-1})} \approx (10^{-2})$$

$$K\nabla \psi \approx K \frac{\psi}{L} \approx \frac{(0.2)(0.45)}{(2.5x10^{-1})} \approx (10^{-1})$$

$$\overline{vq} \approx v_{o1}q \approx \frac{6x10.5}{100} \approx 1.9x10^2 \approx 0(10^2)$$

$$j_a \quad \text{Uncertain} \quad j_c$$
Here the liquid flow velocity, \( v_{ol} \), (arbitrarily taken as 1/100 of the superficial gas velocity) times \( q \) term dominates. In the absence of flow the conductive term dominates. However, the \( J_a \) and \( J_c \) fluxes appear to be equally important as shown by the experimental \( I \) versus \( t \) plots.

It is interesting to examine the boundary condition for steady state conditions. Here \( j = 0 \), and since diffusive transport is slower than conductive transport, Eq. (8) becomes for one-dimensional geometry

\[
D \frac{\partial q}{\partial t} = \frac{KHF}{2 \eta F_\lambda} \left( 1 - \frac{C_s}{C_o} \right) = J_a + J_c ,
\]

where

\[
\lambda = \frac{d}{Nu} \quad \text{(10)}
\]
\[
Nu = \frac{h d}{D} \quad \text{(11)}
\]

and \( C_s \) and \( C_c \) refer to concentrations at the surface and solution bulk, respectively. Here \( \lambda \) is the so-called film thickness, \( d \) is a reference length, \( Nu \) is the Nusselt number, and \( h \) is a charge transfer coefficient. It appears possible that one can employ a procedure similar to that in Ref. 17 in order to deduce the \( h \) dependence on experimental parameters. Presently this is not possible because of lack of a nondimensional correlation for the experimental data and a solution to the charge transport equation.

A well-accepted approach that may be used to correlate the experimental data for the purpose of scaling up to a larger cell is dimensional analysis. Two major drawbacks in the use of this type of procedure are:

1. It assumes a simple product of powers dependence for the dependent variable. In many cases exact solutions do not reduce to simple formulas over wide ranges of experimental conditions.
2. It does not give us a mechanistic interpretation of the reaction or reactions, which would provide a valuable input in the design of a battery for optimum performance.

Nevertheless, it is a powerful tool because, if the variables are properly selected, the number of experiments can be reduced by orders of magnitude through variations in only the resulting non-dimensional groups.

The polarized cell current, $I_{SS}$, was observed to depend upon the following: $A_a$, $A_c$, $l$, $M_a - \chi M_c$, and $FC^0$ where $l$ is the inter-electrode spacing, $M_a$ the rate of dissolution of the anode, $\chi$ a hydrogen blocking factor, $M_c$ a removal rate of hydrogen, and $FC^0$ the initial concentration of salt expressed in C/eq, a measure of the salinity. Other parameters which are important are the fluid viscosity, $\mu$, density, $\rho$, and diffusion coefficient, $D$. Following the procedure outlined in Ref. 18 we obtain the following matrix in the MLTQ (ma$^2$s, length, time, charge) system.

$$
\begin{pmatrix}
0 & 0 & 0 & 0 & 0 & 1 & 1 & 0 & 1 & 0 \\
1 & 0 & 2 & 2 & 1 & -1 & -3 & 2 & 0 & -3 \\
1 & -1 & 0 & 0 & 0 & -1 & 0 & -1 & -1 & 0 \\
1 & 1 & 0 & 0 & 0 & 0 & 0 & 0 & 0 & 1 \\
\end{pmatrix}
$$

The rank of the right-hand matrix is nonzero; therefore the number of non-dimensional products is $9-4 = 5$. We obtain the following non-dimensional products:

- 50 -
\[ \pi_1 = \frac{I_{ss}}{FC^0} \]
\[ \pi_2 = \frac{A_a^{1/2}}{L} \]
\[ \pi_3 = \frac{A_c^{1/2}}{L} \]
\[ \pi_4 = \frac{t}{L} \]
\[ \pi_5 = \frac{v}{D} = \text{Sc} \]

where \( \dot{M}_a - \dot{X}_c / \rho D \) is a characteristic length, \( L \)
\[
L = \frac{\dot{M}_a - \dot{X}_c}{\rho D}
\]

and \( \text{Sc} \) is the Schmidt Number for mass transfer. From Buckingham's theorem we obtain

\[
I_{ss} = \phi \left( \frac{(\dot{M}_a - \dot{X}_c)FC^0}{\rho} \right) \left( \frac{A_a}{A_c} \right) f \left( \frac{t}{L} \right) g \text{Sc} \quad (12)
\]

Since \( \dot{M}_a \times t \) for the present study is much less than \( C \) \( \text{Mg}^{++} \) in the salt solution, \( \dot{M}_a \) could not be determined. For example, if we assume a current of 100 mA for 1 hour, then from Faraday's law we obtain

\[
\text{No. of Equivalents of Mg} = \frac{(100 \times 10^{-3})(3600)}{96500} = 3.73 \times 10^{-3} \text{ eq}
\]

or 0.0453 g. This is much less than the value of 0.78 g given in the Appendix for the solution \( C^* = 112 \text{ g/l} \). In future studies, however, \( \text{MgCl}_2 \) could be omitted from the artificial seawater in order to determine \( \dot{M}_a \). No deleterious side effects would be expected in doing so; actually, the rate of diffusion of \( \text{Mg} \) into solution would be enhanced. It is also noted (Ref. 13) that common anion electrolytes such as \( \text{MgBr}_2 \), thus also \( \text{MgCl}_2 \), cause the anode potential to fall rapidly
with increasing pH, while with NaBr, the anode potential was essentially constant over pH 9.3 to 10.0. For a vigorous reaction, $N_c$ can be determined by measuring the rate of evolution of H$_2$ on a gas burette. More sensitive techniques, e.g., use of gas chromatography, would be needed at low H$_2$ concentrations. The blockage factor $\chi$ would be the only difficult parameter to determine since H$_2$ bubble size, rate of formation, and removal would have to be determined as a function of interelectrode spacing and $A_c/A_a$. This would have to be accomplished photographically, and interpretation of the data at small inter-electrode spacings would be a difficult task.

The intent of the discussions under the Charge Transport section was to outline approaches that may be used to obtain an understanding of the transport mechanisms occurring in a seawater battery. It was also emphasized that such an approach is necessary in order to design a battery for optimum performance. The difficulties involved in a program of this type are also outlined. Here the objective is to draw attention to the need for a long-range study program pursuing a number of the ideas presented herein in conjunction with the experimental effort.
7. CONCLUDING REMARKS

There are a number of important conclusions that can be formed from the present experiments. The one of greatest importance, since it pertains to the overall program, is that the development of a high energy density battery subject to the following rigid constraints,

a. Generate 35 kW of continuous power for periods of around 1 hour,

b. Operate within a cylindrical configuration - 21 inches x 7 feet, i.e., 16.8 ft³,

c. Be reliable and have good storage properties, and

d. Be cost effective,

appears feasible. A long range study program is therefore recommended.

The initial phase of this study program could proceed along the lines outlined in the detailed study program in Ref. 19. A few of the more important studies that are needed before scale-up to a practical prototype are

a. Performance characteristics for porous magnesium and lithium composites,

An exact determination of the minimum amount of chlorine needed to sustain optimum battery performance,

c. An investigation of recycling unreacted gases from cell to cell in a small lab-size battery,

d. An exploratory study of other less toxic reacting gases and gas mixtures, and

e. A complementary analytical effort.

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APPENDIX

ESTIMATION OF MASS FLOW RATES

A porous material may be defined as a solid that contains a large number of holes or pores that may be interconnected or distributed randomly or uniformly throughout the solid. The sintered porous stainless steel cathodes are characterized by highly tortuous and interconnected pore paths. In a previous study on controlled gaseous injection in reentry wakes (Ref. 20) it was shown that there are conditions under which these types of porous materials can provide radially uniform injection velocities. In the context of battery development it is of paramount importance to provide a uniform injection velocity because this will have a direct bearing on the electrochemical reaction (a uniform reaction rate across the electrode surface is desirable).

In Ref. 20 the requirement for a radially uniform injection velocity for a porous material exhibiting overall cylindrical geometry was that \( R/\sqrt{K_D} > 10 \). Here \( R \) is the radius of the porous material and \( K_D \) is the permeability, a measure of the ease of flow through a porous medium. The permeability, \( K_D \), is a constant in Darcy's law, which is given by

\[
v_o = \frac{K_D \Delta P}{\mu L},
\]

where \( v_o \) is the superficial velocity, defined as

\[
v_o = v/P.
\]

Here \( v \) is a cross-sectional velocity, and \( P \) is the porosity given by

\[
\%P = \frac{V_p}{V_t} \times 100%.
\]
where \( V_p \) is the pore volume and \( V_t \) is the total volume (solid plus voids). The remaining quantities are \( \mu \), the fluid viscosity, \( \Delta p \), the applied pressure drop, and \( L \), the thickness of the porous material. If the permeability of a material is known, then use of Darcy's law and continuity,

\[
\dot{m} = \rho v_o A_{\text{eff}}
\]

where \( A_{\text{eff}} \) is the cross-sectional area available for flow, allows one to determine \( \dot{m} \), the mass flow rate.

The porosities of the stainless steel samples were determined experimentally using the bulk density technique (Ref. 21). Values of 60.5\%, 46.5\%, and 43\% were obtained for the 165\#, 35\#, and the 5\# samples, respectively. These values are close to the value for cubic packing of spheres, which is 47.6\%. For cubic packing the void diameter is 0.82 times the grain diameter (Ref. 22). In Table 3 values of permeability are calculated based on geometrical properties (porosity and effective pore diameter) alone. The very low values of \( K \) make \( R/K >> 10 \) and, therefore, one can expect a radially uniform injection velocity with these stainless steel samples.

In order to calculate \( v_o \) from Darcy's law, one needs to know the pressure in the electrode holder. Because a minimum of about 2 feet of tygon tubing separated a bourdon pressure gauge from the plenum chamber, \( \Delta p \) was determined by an alternate procedure. The volume of \( N_2 \) injected for similar experimental conditions was measured by displacement of water from an inverted graduated cylinder for different time intervals. The average of seven tests for the volumetric flow rate was 6.6 cc/sec. Now, assuming the same volumetric flow rate for chlorine and standard conditions, we obtain

\[
\dot{m} = 6.6 \text{ cc/sec} \times \frac{70.9 \text{ g} \text{ Cl}_2}{22400 \text{ cc}} \times \frac{1 \text{ lb}}{453 \text{ g}} = 4.6 \times 10^{-5} \text{ lb/sec}
\]
Table 3
Permeability Based on Geometrical Properties

<table>
<thead>
<tr>
<th>Model</th>
<th>Permeability Equation</th>
<th>Permeability for Stainless Steel Samples</th>
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</thead>
<tbody>
<tr>
<td>Serial</td>
<td>$K_D = \frac{1}{96} Pd^2$</td>
<td>$135\mu - 1.24 \times 10^{-9}$ ft$^2$</td>
</tr>
<tr>
<td></td>
<td></td>
<td>$29\mu - 9.03 \times 10^{-9}$</td>
</tr>
<tr>
<td></td>
<td></td>
<td>$4\mu - 7.68 \times 10^{-13}$</td>
</tr>
<tr>
<td>Kozeny Equation*</td>
<td>$K_D = c/16 Pd^2$</td>
<td>$135\mu - 3.70 \times 10^{-9}$</td>
</tr>
<tr>
<td></td>
<td></td>
<td>$29\mu - 1.31 \times 10^{-10}$</td>
</tr>
<tr>
<td></td>
<td></td>
<td>$4\mu - 2.30 \times 10^{-12}$</td>
</tr>
<tr>
<td>Ideal Medium</td>
<td>$K_D = (1/32)Pd^2$</td>
<td>$135\mu - 3.70 \times 10^{-9}$</td>
</tr>
<tr>
<td></td>
<td></td>
<td>$29\mu - 1.31 \times 10^{-10}$</td>
</tr>
<tr>
<td></td>
<td></td>
<td>$4\mu - 2.30 \times 10^{-12}$</td>
</tr>
<tr>
<td>Brinkman Drag Theory</td>
<td>$K_D = \frac{d^2}{72} \left{ 3 + \frac{4}{1-p} - 3 \sqrt{\frac{8}{1-p} - 3} \right}$</td>
<td>$135\mu - 1.69 \times 10^{-9}$</td>
</tr>
<tr>
<td></td>
<td></td>
<td>$29\mu - 1.88 \times 10^{-11}$</td>
</tr>
<tr>
<td></td>
<td></td>
<td>$4\mu - 1.07 \times 10^{-13}$</td>
</tr>
</tbody>
</table>

*Kozeny constant taken equal to 0.5, its value for a circle.

Here the pore diameters are the respective void diameters.
If we use the entire cross-sectional area of the cylindrical configuration \( A = 2.4 \text{ ft}^2 \) for an electrode, then over a 1-hour time period

\[
m_2 \times t = \frac{A_2}{A_1} t = \frac{(4.6 \times 10^{-5})(2.4)(3600)}{\pi/4 (1.672 \times \frac{1}{12})^2} = 26 \text{ lb Cl}_2.
\]

Based on the density of liquid \( \text{Cl}_2 \) at 20°C of 85 lb/ft\(^3\), the volume occupied is only 0.3 ft\(^3\). However, 63 cells are needed to provide 35 kW. If each cell requires 0.3 ft\(^3\) of \( \text{Cl}_2 \), then 19.9 ft\(^3\) of \( \text{Cl}_2 \) would be needed for the 35 kW battery.

Additional tests were carried out to determine the amount of unreacted \( \text{Cl}_2 \), or the amount of \( \text{Cl}_2 \) available for recycling from cell to cell in a battery. The results of two successive tests indicated that around a factor of two was available for recycle. This was obtained by passing \( \text{Cl}_2 \) through the seawater cell with and without the Mg anode in its electrode holder and recording the time required to fill an evacuated chamber from 30 psig (vacuum) to atmospheric. The chamber filled about twice as fast without the Mg anode. Thus, a more realistic value for the 35 kW battery would be around 10 ft\(^3\) instead of the above 19.9 ft\(^3\). It is believed that the minimum amount of \( \text{Cl}_2 \) needed for efficient cell reaction is somewhat less than \( \dot{m} = 4.6 \times 10^{-5} \text{ lb/sec} \). This decrease is expected to offset an increase in \( \text{Cl}_2 \) required for operating below the ideal 100% recycle efficiency. More precise tests using choked orifices would be needed to determine these quantities.

**ESTIMATION OF \( E^0 \) FOR AZ31B-0 MAGNESIUM ALLOY**

Standard materials handbooks give the following weight percentages for the AZ31B-0 magnesium alloy:
Al 2.5 - 3.5%
Zn 0.7 - 1.3%
Mn ~ 0.2%

Assuming that the $E^0$'s add as $\sum X_i \ E^0_i$ where $X_i$ is the mean weight fraction of the respective metals and $E^0_i$ is the difference between the standard anode $E_0$'s and the $E_0$ for the cathode reaction,

$$2 \ H_2O + 2e^- \rightarrow 2OH^- + H_2 - 0.828 \ V,$$

we obtain the following value for the weighted summation of $X_i E^0_i$:

<table>
<thead>
<tr>
<th>Element</th>
<th>$E_0$ (volts)</th>
<th>$E^0_i$</th>
<th>$X_i$</th>
<th>$X_i \ E^0_i$</th>
</tr>
</thead>
<tbody>
<tr>
<td>Al-Al+++3e^-</td>
<td>1.66</td>
<td>0.832</td>
<td>3%</td>
<td>0.0025</td>
</tr>
<tr>
<td>Mn-Mn+++2e^-</td>
<td>1.18</td>
<td>0.352</td>
<td>0.2%</td>
<td>0.0007</td>
</tr>
<tr>
<td>Zn-Zn+++2e^-</td>
<td>0.763</td>
<td>-0.065</td>
<td>1.0%</td>
<td>-0.00065</td>
</tr>
<tr>
<td>Mg-Mg+++2e^-</td>
<td>2.37</td>
<td>1.542</td>
<td>95.8%</td>
<td>1.477</td>
</tr>
</tbody>
</table>

$$\sum X_i E^0_i = 1.48 \ \text{volts}$$

This is not much different from the 1.542 volts for the $E^0$ obtained with pure Mg. However differences will arise in the current because of the greater ease of reaction for pure Mg, e.g., pure Mg will react in air whereas the AZ31B-O Mg alloy requires a strong salt solution to undergo appreciable reaction.
NUMBER OF EQUIVALENTS IN SOLUTION FOR $C^* = 112 \text{ g/l}$

1. $\text{NaCl} = 85.57 \text{ g } 10^3 \text{ cc} \times \frac{1 \text{ eq}}{58.45 \text{ g}} = 1.47 \times 10^{-3} \text{ eq cc}$

$\text{Na}_2\text{SO}_4 = 8.67 \text{ g } 10^3 \text{ cc} \times \frac{2 \text{ eq}}{142.03 \text{ g}} = 1.22 \times 10^{-4}$

$\text{MgCl}_2 = 17.45 \text{ g } 10^3 \text{ cc} \times \frac{2 \text{ eq}}{95.23 \text{ g}} = 3.66 \times 10^{-4}$

$\text{KCl} = 2.65 \text{ g } 10^3 \text{ cc} \times \frac{1 \text{ eq}}{74.56 \text{ g}} = 3.55 \times 10^{-5}$

$\text{CaSO}_4 = 4.73 \text{ g } 10^3 \text{ cc} \times \frac{2 \text{ eq}}{136.13 \text{ g}} = 9.47 \times 10^{-5}$

Total No. Equivalents $2.1 \times 10^{-3} \text{ eq cc}$

2. Number of grams of Mg present in the reaction cell:

$$\text{grams of Mg} = \frac{17.45 \text{ g } \text{MgCl}_2}{2 	imes 175 \text{ cc} \times \frac{24.32 \text{ g Mg}}{95.23 \text{ g } \text{MgCl}_2}} = 0.78 \text{ g Mg}$$
REFERENCES


10. B.J. Wilson, Characteristics of an Inert Cathode/Magnesium Anode Seawater Battery, NRL 6715, Naval Research Laboratory, Washington, D.C., June 1968.


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